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Electronic and transport properties of carbon and boron-nitride ferrocene nanopeapods

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Electronic and transport properties of novel ferrocene based carbon nanotube (CNT) and boron-nitride nanotube (BNNT) nanopeapods, including $\text{Fe}(\text{Cp})_2@ \text{CNT}$, $\text{Fe}_2(\text{Cp})_3@ \text{CNT}$, $\text{Fe}(\text{Cp})_2@ \text{BNNT}$, and $\text{Fe}_2(\text{Cp})_3@ \text{BNNT}$ (where Cp refers as cyclopentadiene), are investigated using the density functional theory and non-equilibrium Green's function methods. Computed electronic structures of the $\text{Fe}(\text{Cp})_2@ \text{CNT}$ and $\text{Fe}_2(\text{Cp})_3@ \text{CNT}$ nanopeapods suggest that their electric conductivity is primarily contributed by the CNT π channel while the electron hopping from the core $\text{Fe}(\text{Cp})_2$ or $\text{Fe}_2(\text{Cp})_3$ to the sheath CNT may have some contribution to the transport properties. Encapsulating $\text{Fe}(\text{Cp})_2$ into BNNT is more favorable for the electron conduction, owing to the splitting of the BNNT bandgap by the $\text{Fe}(\text{Cp})_2$ state. In contrast, introducing $\text{Fe}_2(\text{Cp})_3$ into the BNNT is not beneficial to the conduction due to intramolecular electron transfer within the core $\text{Fe}_2(\text{Cp})_3$ which can cause a trap effect. Because the transport channels can be changed by the applied bias voltage, the transport properties cannot be solely predicted from the electronic structures of infinite systems alone. For computing transport properties, we use two-probe device model systems with a finite-sized nanopeapod sandwiched between two CNT electrodes. Again, we find that encapsulating either $\text{Fe}(\text{Cp})_2$ or $\text{Fe}_2(\text{Cp})_3$ into CNTs has little effect on the conductivity owing to the strong metallic character of the CNT sheath. Encapsulating $\text{Fe}(\text{Cp})_2$ into BNNTs can notably enhance electron conducting due to electron hopping from the core $\text{Fe}(\text{Cp})_2$ to the sheath BNNT. Encapsulating $\text{Fe}_2(\text{Cp})_3$ into BNNTs, however, has little effect on the electron conductivity of BNNT nanopeapods due to the trap effect of the longer guest molecules. Hence, the length of guest molecules can effectively tune electronic and transport properties of the BNNT nanopeapods.

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1 Introduction

Carbon nanotubes are capable of encapsulating guest atoms or molecules into their inner cylindrical space, thus forming a quasi-one-dimensional core-sheath structure of $\text{X}@ \text{CNT}$, where X denotes the core atoms or molecules. Thus far, a large number of organic and inorganic species, such as fullerenes,^{1–6} single elements,^{7,8} multimeric thiophene oligomers,^{9,10} methyl chloride,¹¹ oxides,¹² *etc.* have already been introduced into CNT's cylindrical inner space as guest molecules, thereby open exciting possibility of creating new materials that can be tailored to a particular electronic functionality. For example, the iron-fullerene complexes ($\eta^5\text{-Cp}$) $\text{FeC}_{60}\text{R}_5$ (R = Me and Ph; Cp = C_5H_5) have been successfully encapsulated into CNTs by Nakamura *et al.*¹³ Li *et al.* demonstrated “ferrocene nano-peapods” by cramming ferrocene molecules into the CNTs.¹⁴

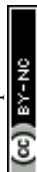
Boron nitride nanotubes (BNNTs)^{15,16} are another prevailing tubular nanomaterials which have also been the subject of

extensive studies over the past decade. Like CNTs, BNNTs also possess many intriguing properties such as strong hardness, high thermal conductivity and chemical inertness.^{17–19} Indeed, BNNTs are known to be more chemically and thermally stable than CNTs.²⁰ As such, BNNTs are especially suitable for the task of shielding and protecting guest molecules from external chemical attack.²¹ Many previous studies have reported that BNNTs can be used to encapsulate metal nanowires (Ni, Co, Fe, and Cu),^{22–26} oxides ($\alpha\text{-Al}_2\text{O}_3$ and $\text{Al}_{18}\text{B}_4\text{O}_{33}$),^{27,28} SiC carbides,^{29,30} GaN nitrides,^{31,32} potassium halide nanowires,³³ or fullerenes.³⁴

Recently, the potential to change electron transport properties of CNTs and BNNTs by encapsulating suitable atoms or molecules has attracted increasing attention from researchers in many fields. However, the specific role of guest molecules inside nanotubes on the electronic and transport properties can be highly system-dependent. For example, a previous study shows that the electric conduction pathway within $\text{Fe}@ \text{CNT}$ is actually through CNT percolation³⁵ while the encapsulated iron nanoparticles have little influence on the CNT conductivity. However, Hsu and coworkers found that the Fe core in CNTs can generate an electromagnetic inductive phase, suggesting some participation in electron transport.³⁶ Therefore, a theoretical study of the one-dimensional (1D) core-sheath

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structures $X@CNT$ and $X@BNNT$ will be informative for understanding special effects of the core on the electronic and transport properties of endohedral $X@CNT$ and $X@BNNT$ materials.

It is known that CNTs can be either metallic or semiconducting, depending on their diameter and chirality, while the BNNTs are always insulating with bandgaps of 4–5 eV.^{37–40} Such a striking difference in electronic properties between CNTs and BNNTs will inevitably lead to different physical properties of the corresponding endohedral $X@CNT$ and $X@BNNT$. For example, density-functional theory (DFT) calculation suggests that the $(\eta^6-C_{60}-V)@CNT$ nanopeapod is metallic with characteristics of multiple carriers contributed from the CNT, C_{60} , and V, while the $(\eta^6-C_{60}-V)@BNNT$ nanopeapod is predicted to be semiconducting with a narrow bandgap, and its charge carriers are contributed only by the $C_{60}V$ chain.⁴¹

In the laboratory, many multidecker organometallic sandwich clusters have been synthesized.^{42,43} It has been shown that the 1D sandwich compounds can exhibit semiconducting or even conducting properties.^{44,45} The $Fe_n(Cp)_{n+1}$ nanowires have attracted much attention as they are promising components for nanoelectronic devices, especially in high density storage and quantum computing.⁴⁶ The $Fe_n(Cp)_{n+1}$ sandwich clusters have been synthesized in the gas phase and characterized by mass spectroscopy.⁴² Note that ferrocene molecules have already been encapsulated into CNTs to create a class of self-assembled hybrid structures named as “ferrocene nanopeapods”.¹⁴ Thus, we expect that the $Fe_n(Cp)_{n+1}$ clusters may also be encapsulated into nanotubes to form “ $Fe_n(Cp)_{n+1}$ nanopeapods”. In this study, we investigate the electronic and transport properties of $Fe_n(Cp)_{n+1}$ nanopeapods and their differences from pure nanotubes. Specifically we select $Fe(Cp)_2@CNT$ (**1a**), $Fe_2(Cp)_3@CNT$ (**1b**), $Fe(Cp)_2@BNNT$ (**2a**), and $Fe_2(Cp)_3@BNNT$ (**2b**) as prototype nanopeapod systems to compute their electronic structures and transport properties using DFT and non-

equilibrium Green's function (NEGF) methods. We find that the length of the core per supercell of CNTs, *i.e.*, $Fe(Cp)_2$ versus $Fe_2(Cp)_3$, can have notable effects on the electronic and transport properties of the CNT and BNNT nanopeapods.

2 Models and computational methods

Fig. 1 shows optimized structures of nanopeapods **1a**, **1b**, **2a**, and **2b**. Here, metallic CNTs (6, 6) and insulating BNNTs (6, 6) are selected as the host nanotubes. For computing electronic structures, the infinite nanotube systems of **1a**, **1b**, **2a**, and **2b** are modeled *via* using the periodic conditions in the axial direction. For computing transport properties, the two-probe devices (a unit cell of the nanopeapod sandwiched between two CNT (6, 6) electrodes) are adopted. For the purpose of the benchmark test, we have computed electronic properties using infinite nanotube systems and transport properties using the two-probe devices for pure CNTs (6, 6) and BNNTs (6, 6).

For the periodic systems, the supercell contains 120 atoms for the nanotubes and one $Fe(Cp)_2$ or $Fe_2(Cp)_3$ unit inside the nanotube. The Cps of $Fe(Cp)_2/Fe_2(Cp)_3$ are packed along the axial direction as shown in Fig. 1. The supercell length L is about 12.3 Å in the axial direction (z direction), long enough to neglect intermolecular interactions among core molecules $Fe(Cp)_2$ or $Fe_2(Cp)_3$. The nanopeapods are separated by 20.0 Å in two other directions to neglect inter-tube interactions. All the periodic systems are fully optimized until the maximum absolute force is less than $0.02 \text{ eV } \text{\AA}^{-1}$.

For the two-probe systems, we carve out a supercell for each nanopeapod system as the central scatter region which is sandwiched between two identical CNT (6, 6) electrodes (*cf.* **1a** and **2a** in Fig. 1). A reason for choosing CNTs (6, 6) as the electrodes is its known metallic character and good matching

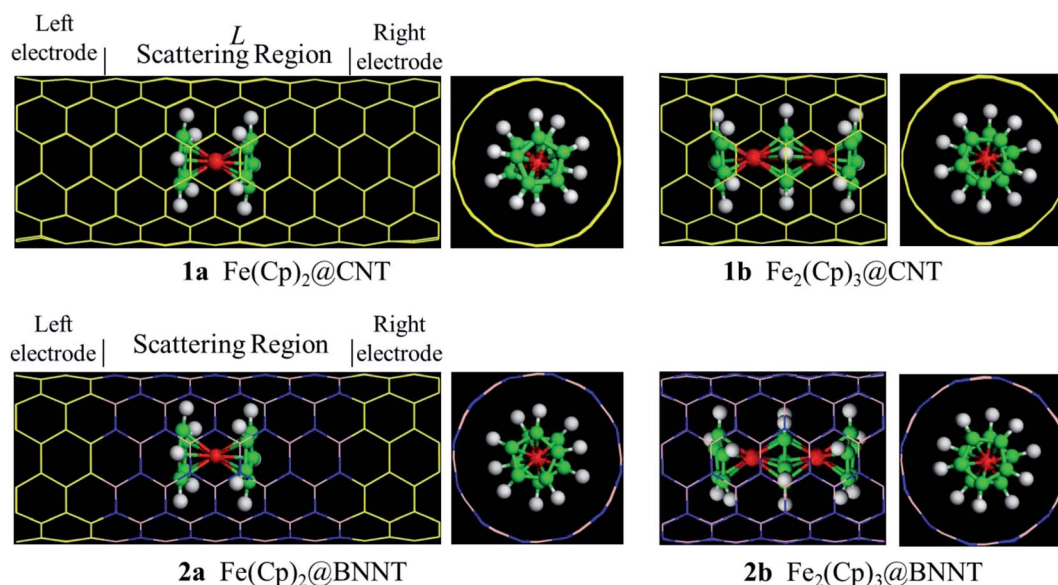


Fig. 1 Optimized structures of **1a**, **1b**, **2a**, and **2b**. **1a** and **2a** also illustrate the two-probe devices for electron transport computation.



with the scatter region of peapods. The entire two-probe devices are denoted as CNT/peapod/CNT, *i.e.*, CNT/**1a**/CNT (CNT/Fe(Cp)₂@CNT/CNT), CNT/**1b**/CNT (CNT/Fe₂(Cp)₃@CNT/CNT), CNT/**2a**/CNT (CNT/Fe(Cp)₂@BNNT/CNT), and CNT/**2b**/CNT (CNT/Fe₂(Cp)₃@BNNT/CNT). The entire system including two electrodes and a scatter region is fully optimized until the maximum absolute force is less than 0.02 eV Å⁻¹. Transport current is computed by changing the applied bias in the step of 0.2 V in the range of -1.0 to 1.0 V.

All the computations for the infinitely long and two-probe systems are performed using an *ab initio* code package, Atomistix ToolKit (ATK), which is based on the combination of DFT and the NEGF methods.^{47–50} A generalized gradient approximation (GGA) within the Perdew–Burke–Ernzerhof (PBE) formalism is employed to describe the exchange correlations between electrons. A double- ζ basis functional with polarization (DZP) is used for all atoms. A (1 × 1 × 150) *k*-point in the string Brillouin zone (*x*, *y*, and *z* directions, respectively) is used. 150 Ry cutoff energy is applied to describe the periodic wave function.

3 Results and discussion

First, we investigate geometrics and band structures of the infinitely long nanopeapods of **1a**, **2a**, **1b**, and **2b**, followed by computing transport properties based on the two-probe devices of CNT/nanopeapod/CNT with the metallic CNT (6, 6) electrodes.

3.1 Geometric structures and electronic properties

Stability and geometry. Computed total energies per supercell for the optimized **1a**, **1b**, **2a**, and **2b** are listed in Table 1. For the **1b** and **2b** nanopeapods, we consider two spin states of the two Fe atoms, *i.e.*, the antiferromagnetic (AFM) state and the ferromagnetic (FM) state. It is noteworthy that both **1b** and **2b** favor the FM ground state, indicating the ferromagnetic Fe–Fe coupling. Hereafter, we mainly focus on the FM state for **1b** and **2b** and associated electronic and transport properties.

Chemical stability for encapsulating either Fe(Cp)₂ or Fe₂(Cp)₃ into nanotubes is evaluated by computing the reaction energy per supercell for the net reaction [nanotube (NT) + Fe(Cp)₂ or Fe₂(Cp)₃ → peapod – Δ*E*_r]. Here, the computed reaction energies Δ*E*_r are -3.27, -7.18, -1.89, and -2.93 eV for

1a, **1b**, **2a**, and **2b**, respectively. The negative values indicate exothermic energies. Hence, incorporation of Fe(Cp)₂ or Fe₂(Cp)₃ into either CNTs or BNNTs is energetically favorable.

The optimized supercell length in the axial direction (*L*), the radii of the nanotube (*R*), the face-to-face distances between adjacent Cps (*r*₁), the average C–C bond lengths in Cp (*r*₂), and the distances between the H atom of Cp and the nearest atom of the nanotube (*r*₃) for **1a**, **1b**, **2a** and **2b** are given in Table 2. Incorporation of Fe(Cp)₂ or Fe₂(Cp)₃ into nanotubes induces a slight expansion of the host nanotube as reflected from the larger values of *R* compared to the pristine nanotube. The distances *r*₁ are within the range of 3.188–3.280 Å, very close to the Cp-to-Cp separation in the Fe(Cp)₂ molecule (3.32 Å)^{51–53} but slightly shorter than that (3.40 Å) in the ferrocene dimer.⁵⁴ The computed C–C bond lengths in Cp *r*₂ are in the range of 1.447–1.463 Å for both CNT and BNNT nanopeapods, very close to the experimentally measured bond lengths for ferrocene (1.440 Å).⁵⁵ The shortest distances between the H atom of Cp and the nearest atom of the nanotube (*r*₃) are in the range of 1.998–2.094 Å. Similar to the ferrocene, the H atoms on the terminal Cp rings tilt inward (toward the Fe atoms) by 6–9°.

Electronic band structures. Computed band structures and the projected density of states (PDOS) for the nanopeapods **1a**, **1b**, **2a**, and **2b** are given in Fig. 2–4. For the purpose of comparison, the results of pure CNTs and BNNTs are also given in Fig. 2–4. It is known that for the ferrocene molecule Fe(Cp)₂, the ligand field splitting of the Fe *d* orbitals results in five molecular orbitals mainly consisting of Fe *d*_{xy}, *d*_{x²–y²}, *d*_{z²}, *d*_{xz}, and *d*_{yz}, where the former three are occupied while the latter two are unoccupied.

Fig. 2 displays the band structures of the CNT nanopeapods **1a** and **1b**, as well as the Kohn–Sham orbitals near the Fermi level (*E*_f). For pure CNTs, the valence band I (red line in Fig. 2(a)) and the conduction band II (blue line in Fig. 2(a)) cross at *E*_f with large dispersion, suggesting a typical metallic character. In **1a**, two nearly degenerate and flat levels, the Fe *d*_{xy} and *d*_{x²–y²} bands (cyan and wine lines in Fig. 2(b)), are located just below *E*_f. The Kohn–Sham orbital shows that the Fe *d*_{z²} state is coupled with the Cp π state (denoted as a *d*_{z²}(Fe)– π (Cp) state). Notably, this *d*_{z²}(Fe)– π (Cp) state hybridizes with the CNT π state, giving rise to a band just below *E*_f with modest dispersion (dark yellow line in Fig. 2(b)). Hence, the core–sheath interaction occurs when Fe(Cp)₂ is encapsulated into the CNT. The CNT band I (red

Table 1 Optimized total energies of unit cells of peapods **1a**, **1b**, **2a**, and **2b**

Species	Total energies (in unit of eV)
1a Fe(Cp) ₂ @CNT	–21 307.81607
1b Fe ₂ (Cp) ₃ @CNT (FM)	–22 925.03092
1b Fe ₂ (Cp) ₃ @CNT (AFM)	–22 924.98010
2a (Cp) ₂ @BNNT	–23 612.79319
2b Fe ₂ (Cp) ₃ @BNNT (FM)	–25 267.18630
2b Fe ₂ (Cp) ₃ @BNNT (AFM)	–25 267.11294

Table 2 Computed geometric parameters for **1a**, **1b**, **2a**, and **2b**. Those for the pure CNT and BNNT are listed for comparison (in unit of Å)

Species	<i>L</i>	<i>R</i>	<i>r</i> ₁	<i>r</i> ₂	<i>r</i> ₃
Pure CNT	12.310	8.298			
1a Fe(Cp) ₂ @CNT	12.302	8.352	3.280	1.447	2.016
1b Fe ₂ (Cp) ₃ @CNT	12.313	8.328	3.258	1.458	1.998
Pure BNNT	12.315	8.147			
2a Fe(Cp) ₂ @BNNT	12.304	8.452	3.188	1.452	2.094
2b Fe ₂ (Cp) ₃ @BNNT	12.313	8.489	3.248	1.463	2.078



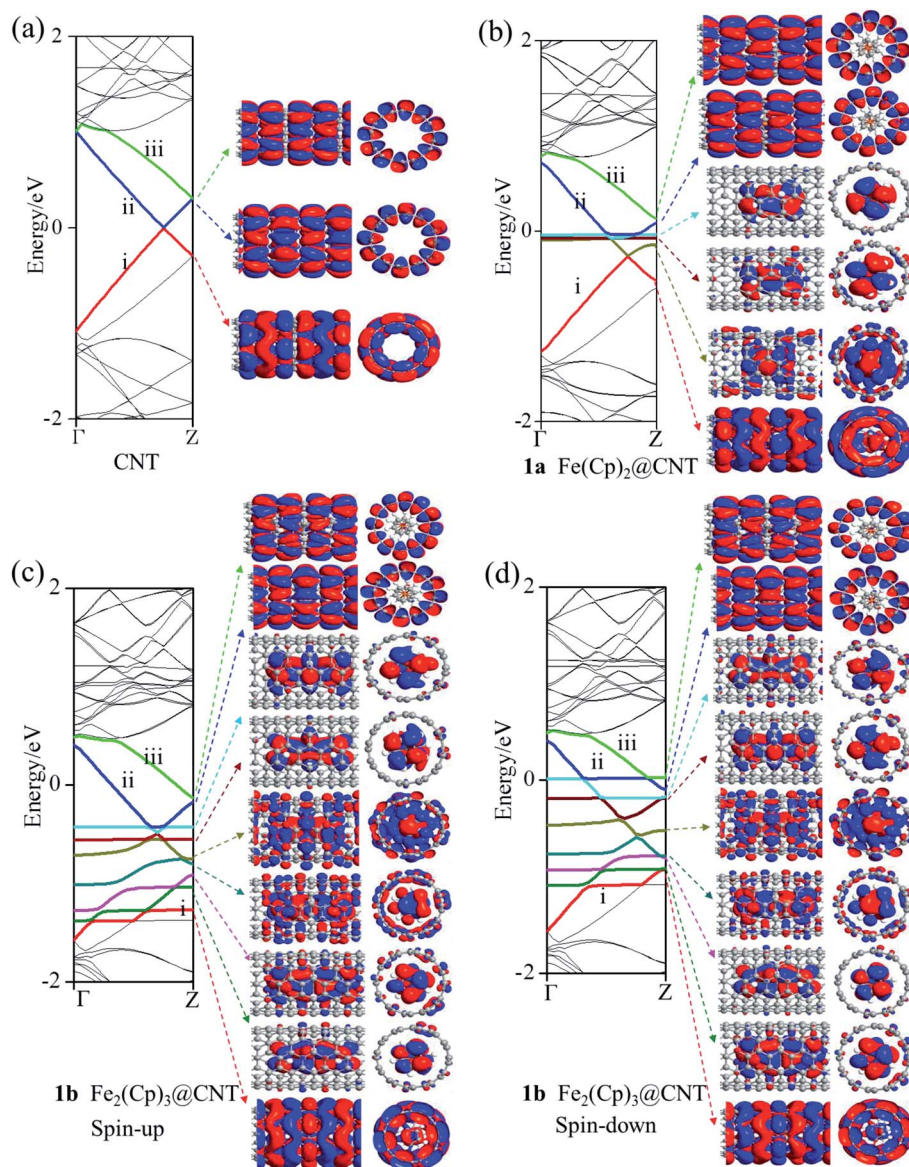


Fig. 2 Computed band structures (left panels) of (a) the pure CNT, (b) **1a**, and (c and d) **1b** nanopeapod systems and the Kohn–Sham orbitals (middle and right panels) corresponding to the energy levels (highlighted in color lines) near E_f at the Γ point. The iso-surface value is $0.05 \text{ (e } \text{\AA}^{-3})$.

line in Fig. 2(b)) is located just below the $d_{z^2}(\text{Fe})-\pi(\text{Cp})$ state and crosses with the $d_{z^2}(\text{Fe})-\pi(\text{Cp})$ state. The unoccupied Fe d_{xz} and d_{yz} states are far away from E_f and hence not displayed in Fig. 2(b). The conduction band still includes the CNT band II in **1a** (blue line in Fig. 2(b)). Thus, there may be two conducting pathways in **1a**: one through the CNT and another through electron hopping from the core $\text{Fe}(\text{Cp})_2$ to the sheath CNT.

For nanopeapod **1b**, the spin-up state clearly shows different features from the spin-down state, which results in a magnetic moment of $0.81 \mu_B$ per supercell (with two Fe atoms), close to $1.0 \mu_B$ per supercell for the $\text{Fe}_n(\text{Cp})_{n+1}$ nanowire.⁵⁶ The Kohn–Sham orbitals plotted in Fig. 2(c) and (d) show the strong core–sheath interaction in **1b** as in **1a**. More bands, originated mainly from the Fe d_{xy} , $d_{x^2-y^2}$, and d_{z^2} states, are introduced below E_f in both spin-up and spin-down states. Again, the

unoccupied Fe d_{xz} and d_{yz} orbitals are far away from E_f and thus not plotted in Fig. 2(c) and (d). Compared to pure CNTs, one can see that the CNT bands II and III (blue and green lines in Fig. 2(c)) in the spin-up state of **1b** downshift considerably, and both bestride E_f . As such, the CNT π state in the spin-up state behaves like half-filled and dominates the electron transport. In the spin-down state, the CNT band II (blue line in Fig. 2(d)) crosses E_f and the CNT band III (green line in Fig. 2(d)) is located just above E_f . Therefore, the CNT π state in the spin-down state still serves as a major transport pathway. One occupied band originated mainly from the Fe d orbital of the spin-down state upshifts across E_f , turned into a half-filled band, which suggests that electrons could transfer to the core $\text{Fe}_2(\text{Cp})_3$. Hence, the core $\text{Fe}_2(\text{Cp})_3$ may also serve as a localized trap-state in the process of electron transport, which is not beneficial to the conducting. Overall, the conductivity of **1b** is



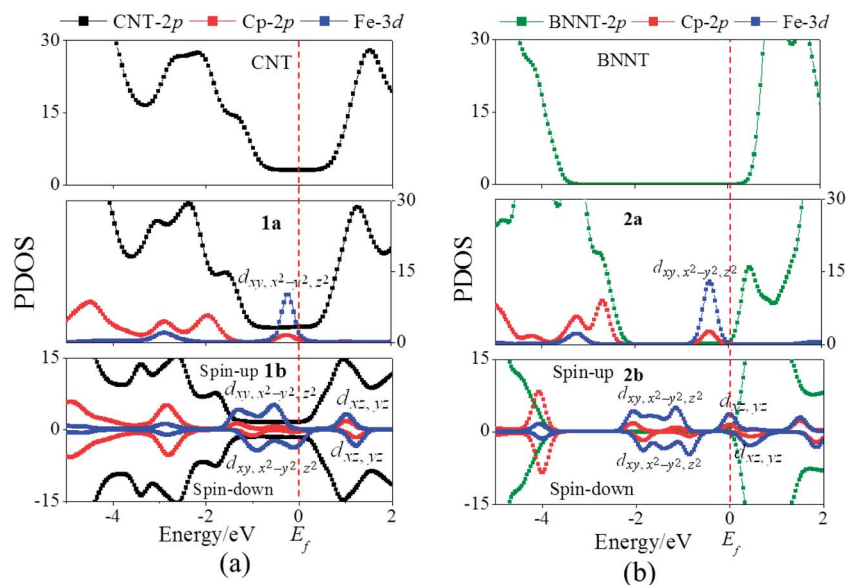


Fig. 3 Computed projected density of states (PDOS) of **1a**, **1b**, **2a**, and **2b** nanopeapod systems. For comparison, PDOS of the pure CNT and BNNT systems are also presented.

likely contributed from multiple factors: (1) the CNT π channel dominates the conductivity; (2) electron hopping from core $\text{Fe}_2(\text{Cp})_3$ to CNTs may have appreciable effects on the transport properties; and (3) $\text{Fe}_2(\text{Cp})_3$ may entails a trap effect on the electron transport.

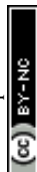
Fig. 3(a) plots the PDOS of pure CNTs, **1a**, and **1b**. Evidently, the CNT π states of pure CNTs, **1a**, and **1b** exhibit similar PDOS peaks and broad valley feature around E_f . Thus, the CNT π states are still the main transport channels even with the core $\text{Fe}(\text{Cp})_2$ or $\text{Fe}_2(\text{Cp})_3$. In **1a**, the PDOS peaks of Fe d_{xy} , $d_{x^2-y^2}$, and d_{z^2} states are located below E_f , hybridized with the CNT π state. The unoccupied Fe d_{xz} and d_{yz} PDOS peaks do not arise even up to 2.0 eV (Fig. 3(a)). The conduction band of **1a** is dominated by the CNT π state. The core $\text{Fe}(\text{Cp})_2$ may participate in the electron tunneling through electron hopping to the CNT. In the case of **1b**, a similar $\text{Fe}_2(\text{Cp})_3$ –CNT hopping channel also arises in the spin-up state. Compared to **1a**, the PDOS peak of the unoccupied Fe d_{xz} and d_{yz} states moves to the lower energy region, but is still located at about 1.0 eV. The splitting of the spin states of **1b** results in an asymmetric PDOS distribution for the spin-up and spin-down states. The Fe d_{xy} , $d_{x^2-y^2}$, and d_{z^2} orbitals in the spin-down state are shifted toward the higher energy region compared to the spin-up state, so that the Fe d orbital of the spin-down state crosses E_f with a half-filled character, again a manifestation of the trap effect due to the core $\text{Fe}_2(\text{Cp})_3$.

In summary, electronic structures of the **1a** and **1b** systems suggest that the electric conductivity is primarily contributed by the CNT π channel, while electron hopping from the core $\text{Fe}(\text{Cp})_2$ or $\text{Fe}_2(\text{Cp})_3$ to the sheath CNT may have some contribution to the transport properties. On the other hand, the core $\text{Fe}_2(\text{Cp})_3$ may also have a trap effect on the electron transport. Comparison with transport properties obtained based on the

two-probe devices of CNT/**1a**/CNT and CNT/**1b**/CNT will be discussed in Section 3.2.

In Fig. 4, computed band structures of the pure BNNT, **2a**, and **2b** are plotted, so are the Kohn–Sham orbitals near E_f . For pure BNNTs, as expected, the valence band I (red line in Fig. 4(a)) and the conduction band II (green line in Fig. 4(a)) are separated by a large bandgap of 4.52 eV. In **2a**, the core $\text{Fe}(\text{Cp})_2$ introduces three flat bands (cyan, wine, and dark yellow lines in Fig. 4(b)) in the range of -0.37 to -0.43 eV and these bands are mainly contributed from the occupied Fe d_{xy} , $d_{x^2-y^2}$, and d_{z^2} orbitals. The unoccupied orbitals stem mainly from Fe d_{xz} and d_{yz} are far away from E_f and thus not shown in Fig. 4(b). Unlike **1a** which entails strong core–sheath interactions, in **2a** little FeCp_2 –BNNT interaction occurs as reflected from the Kohn–Sham orbitals and the flat feature of the $\text{Fe}(\text{Cp})_2$ bands in Fig. 4(b). The three nearly degenerate $\text{Fe}(\text{Cp})_2$ bands split the native BNNT band gap into two subgaps. As a result, the $\text{Fe}(\text{Cp})_2$ state turns into the valence band and the BNNT band II is the conduction band. As such, the bandgap is reduced to 0.73 eV for **2a** from 4.52 eV of the pristine BNNT. This notable band reduction indicates that electron hopping from the core $\text{Fe}(\text{Cp})_2$ to the sheath BNNT is significant in **2a**.

For **2b**, more orbitals originated from the core $\text{Fe}_2(\text{Cp})_3$ appear in the bandgap region of BNNTs. Clearly, the spin-up and the spin-down states are different, leading to a magnetic moment of $0.98 \mu_B$ per supercell. In the spin-up state, the Fe d_{xz} and d_{yz} orbitals, due to coupling with the Cp π orbital, are downshifted and across E_f (cyan and wine lines in Fig. 4(c)), showing a half-filled character and a trap-state. This case differs from **1a**, **1b**, and **2a** for which the Fe d_{xz} and d_{yz} orbitals are located far away from E_f . For the spin-down state, the conduction band is contributed by Fe d_{xz} and d_{yz} orbitals coupled with certain BNNT π orbitals as can be seen from the Kohn–Sham orbitals shown in Fig. 4(d) (cyan and wine lines). The valence



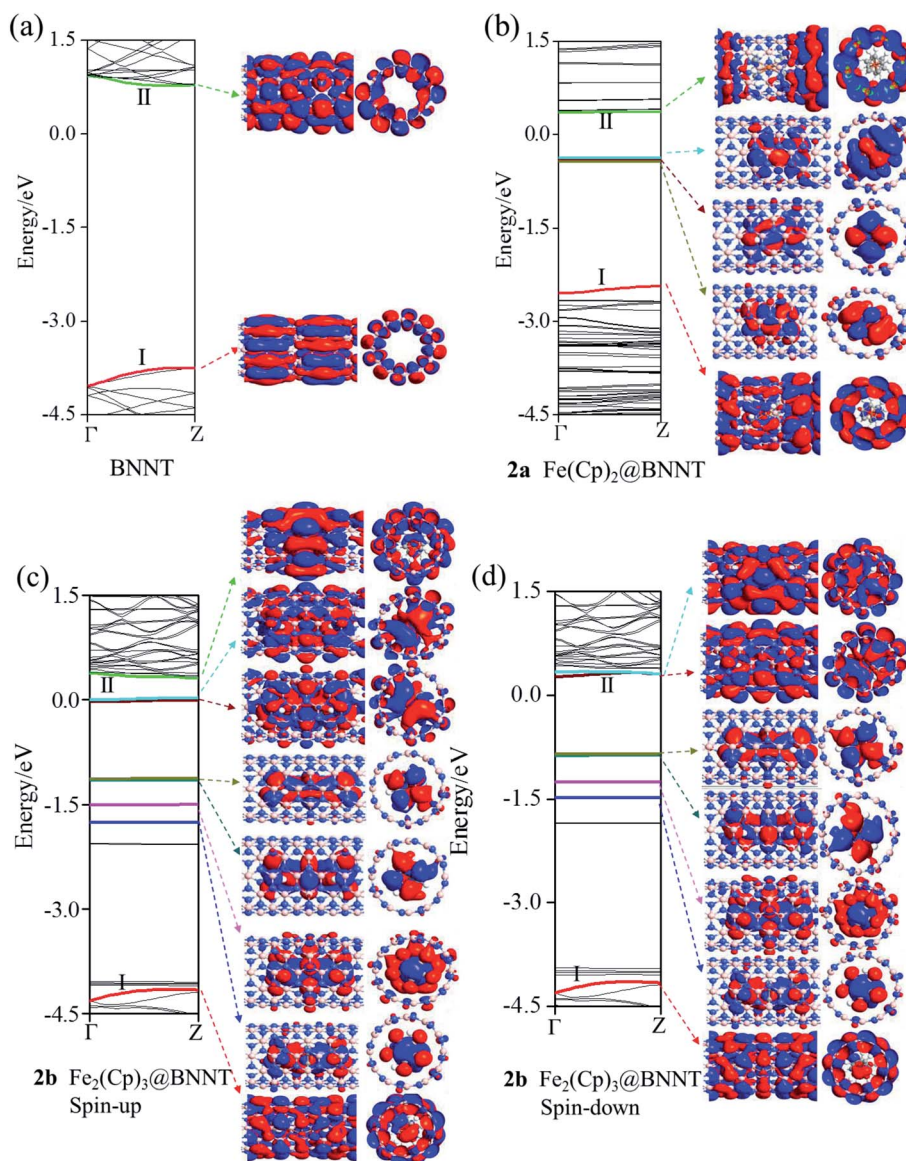


Fig. 4 Computed band structures (left panels) of the pure BNNT, **2a**, and **2b** nanopropod systems and the Kohn–Sham orbitals (middle and right panels) corresponding to the energy levels (highlighted in color lines) near E_f at the Γ point. The iso-surface value is 0.05 ($e \text{ \AA}^{-3}$).

band (dark yellow and dark cyan lines in Fig. 4(d)) stems only from Fe d_{xy} and $d_{x^2-y^2}$ orbitals. Hence, the core $\text{Fe}_2(\text{Cp})_3$ can also have a trap effect on the electron transport in the spin-down state.

Fig. 3(b) shows computed PDOS of the pure BNNT, **2a**, and **2b**. The BNNT retains the insulating properties in **2a** and **2b**, as the BNNT π PDOS are largely separated above and below E_f . In **2a**, the Fe d_{xy} , $d_{x^2-y^2}$, and d_{z^2} orbitals contribute a peak below E_f , resulting in a valence state, while the BNNT π orbitals also contribute to the conduction state. Thus, electrons can be transported by hopping from the core $\text{Fe}(\text{Cp})_2$ to the sheath BNNT, consistent with the conclusion based on computed band structures. Like in **1a**, the unoccupied Fe d_{xz} and d_{yz} PDOS are far away from E_f (Fig. 3(b)). No PDOS hybridization between the BNNT and the FeCp_2 is seen near E_f , suggesting little $\text{Fe}(\text{Cp})_2$ –BNNT interaction in **2a**. This result is consistent with

the Kohn–Sham orbital diagram in Fig. 4(b). In **2b**, the Fe d_{xz} and d_{yz} orbitals, coupled with the Cp π orbital, are downshifted less substantially compared with **1a**, **1b**, and **2a**. As a result, the Fe d_{xz} and d_{yz} PDOS peaks appear just across E_f in the spin-up state, showing a half-filled feature and a trap state. Overall, the Fe d PDOS in the spin-up and spin-down states are asymmetric. In the spin-down state, the Fe d_{xz} and d_{yz} orbitals are located just above E_f while hybridized with the BNNT π state, resulting in the conduction band; the valence band originates from the Fe d_{xy} , $d_{x^2-y^2}$, and d_{z^2} orbitals. Therefore, the trap effect could also exist in the spin-down state. These results are in line with the band structure analysis.

From the above analysis for **2a** and **2b**, it seems that encapsulating $\text{Fe}(\text{Cp})_2$ is more favorable for the electron conduction, owing to the splitting of the BNNT bandgap by the $\text{Fe}(\text{Cp})_2$ state. Electrons could transport through hopping from the core $\text{Fe}(\text{Cp})_2$ to



the sheath BNNT. In contrast, introducing $\text{Fe}_2(\text{Cp})_3$ into the BNNT is not beneficial to the conduction due to intramolecular electron transfer within the core $\text{Fe}_2(\text{Cp})_3$ which can cause a trap effect. More related discussions are given in Section 3.2.

3.2 Two-probe devices

To obtain more quantitative transport properties of the nanopeapods and analyze the effect of core $\text{Fe}(\text{Cp})_2$ or $\text{Fe}_2(\text{Cp})_3$ on the transport properties, we construct a model system such that a supercell of **1a**, **1b**, **2a**, or **2b** is sandwiched between two CNT (6, 6) electrodes to form two-probe devices (denoted as CNT/nanopeapod/CNT). Calculation results based on this model system suggest that electric conductivities of the two-probe devices are mostly consistent with the electronic structures of the corresponding infinitely long nanopeapod systems, although the computed electronic structures can be influenced by applied external bias voltage.

The computed I - V curves based on the two-probe devices are given in Fig. 5. It can be seen that pure CNTs, **1a**, and **1b** show a metallic feature and give nearly the same I - V curves (Fig. 5(a)), suggesting that the conductivity is entirely contributed by the CNT (π state) in the two-probe devices, consistent with the conclusion based on the computed electronic structures. Moreover, conductivities of the CNT/**1b**/CNT and CNT/**2b**/CNT devices appear independent of the spin state within the considered bias range of -1.0 to 1.0 V, although the band structures and PDOS distributions are dependent on the spin state. Fig. 5(b) shows that the pure BNNT, **2a**, and **2b** exhibit features of insulators. Their conductivities follow the sequence **2a** > pure BNNT > **2b**, again consistent with the predication based on computed electronic structures, that is, encapsulation of $\text{Fe}(\text{Cp})_2$ into BNNTs would enhance conductivity while encapsulating $\text{Fe}_2(\text{Cp})_3$ into BNNTs would lower the conductivity.

In general, computed electric conductivity is dependent on multiple factors, including, for examples, energy spectra, the molecular projected self-consistent Hamiltonian (MPSH) states, the PDOS, the transmission spectra (TS), and electrostatic potentials, among others. Next, we analyze the extent to which these factors affect the transport properties of each two-probe

CNT/nanopeapod/CNT devices. Results are depicted in Fig. 6–10, from which one can see that the applied external bias voltage can alter the transport channel.

CNT/1a**/CNT.** As shown in Fig. 6(c), at zero bias ($V = 0$), the Fe d_{xy} , $d_{x^2-y^2}$, and d_{z^2} states contribute to three frontier occupied orbitals, namely, the highest occupied molecular orbital (HOMO), HOMO–1, and HOMO–2. The CNT valence π orbital is located just below the Fe d_{z^2} state and contributes to the HOMO–3. The lowest unoccupied molecular orbital (LUMO) of CNT/**1a**/CNT stems from the unoccupied CNT π orbital. The energy spectrum of CNT/**1a**/CNT is in line with the band structures of the infinite **1a** system. At 1.0 V bias, the CNT valence π orbital (HOMO–3) is shifted upward and becomes the HOMO (Fig. 6(d)), while the LUMO is still originated from the unoccupied CNT π state. This result demonstrates that the external electric field can induce orbital rearrangement of CNT/**1a**/CNT, which can be further reflected from the PDOS in Fig. 7. The Fe d_{xy} , $d_{x^2-y^2}$, and d_{z^2} states yield a sharp peak below E_f at 0.0 V bias voltage (left panel in Fig. 7(a)), while this peak is pushed to the lower energy region and even out of the bias window when the bias $V = 1.0$ V is applied (left panel in Fig. 7(b)). Here, the bias window refers to $[-V/2, V/2]$. Generally, only states within the bias window contribute to the total current. More importantly, the HOMO and LUMO MPSH states spread over the entire CNT, which suggests an effective transport path. Therefore, the conductivity of **1a** is fully determined by the CNT while the core $\text{Fe}(\text{Cp})_2$ plays little role in the conductivity. This is why the I - V curves shown in Fig. 5(a) between pure CNTs and **1a** exhibit little difference. The transmission spectra (left panels in Fig. 8) also show that pure CNTs and **1a** should have the same conductivity as they have almost the same transmission spectral distributions at 1.0 V bias. To evaluate the efficiency of the electron transport through the nanopeapod we compute the molecular electrostatic potentials at 0.0 and 1.0 V bias voltages. The calculation result can also provide atomic-level information about the electron transfer barrier. In any case, the CNT should serve as a major transport path. The application of bias greatly changes the potential distribution due to the external electric field. By

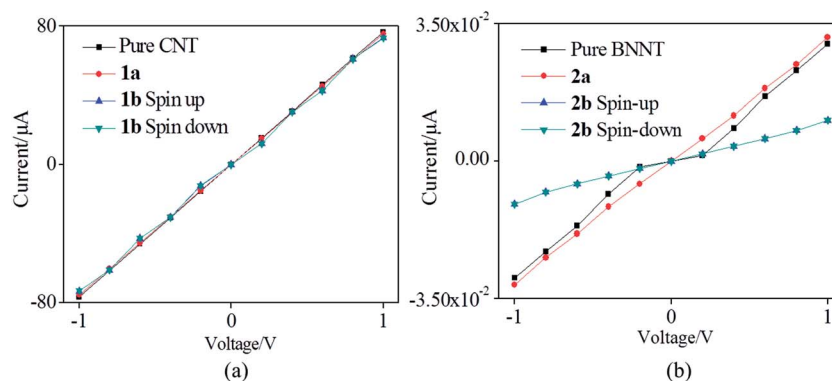


Fig. 5 Computed I - V curves of two-probe devices with finite-sized **1a**, **1b**, **2a**, and **2b** between the two electrodes. For comparison, the I - V curves of the pure CNT and BNNT two-probe devices are also presented.



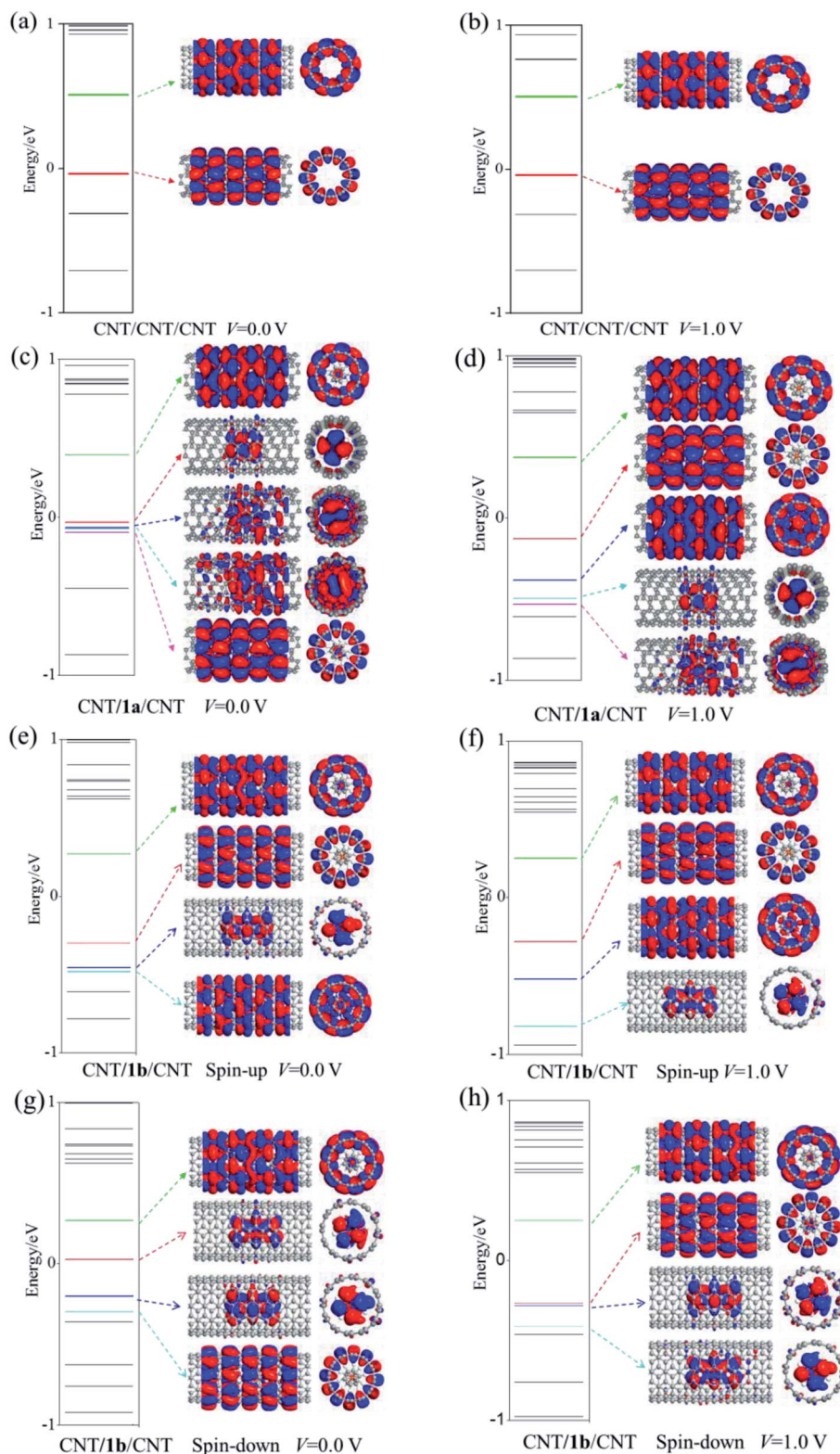


Fig. 6 Computed energy spectra (left panels) of (a and b) the pure CNT; (c and d) **1a**; and (e–h) **1b** two-probe devices. The MPSH states (middle and right panels) correspond to the energy levels (highlighted in color lines) near E_F . The iso-surface value is 0.05 ($e \text{ \AA}^{-3}$).

comparing results of **1a** and those of pure CNTs, one can see that the core $\text{Fe}(\text{Cp})_2$ leads to two major effects. First, it acts as a barrier for the transport (blue part in Fig. 9 for systems $\text{CNT}/\mathbf{1a}/\text{CNT}$ at $V = 1.0 \text{ V}$ bias), and second, it enhances

electron transport (red part in Fig. 9 for systems $\text{CNT}/\mathbf{1a}/\text{CNT}$ at $V = 1.0 \text{ V}$ bias). The charge carriers can effectively permeate from the left electrode to the right electrode as can be seen



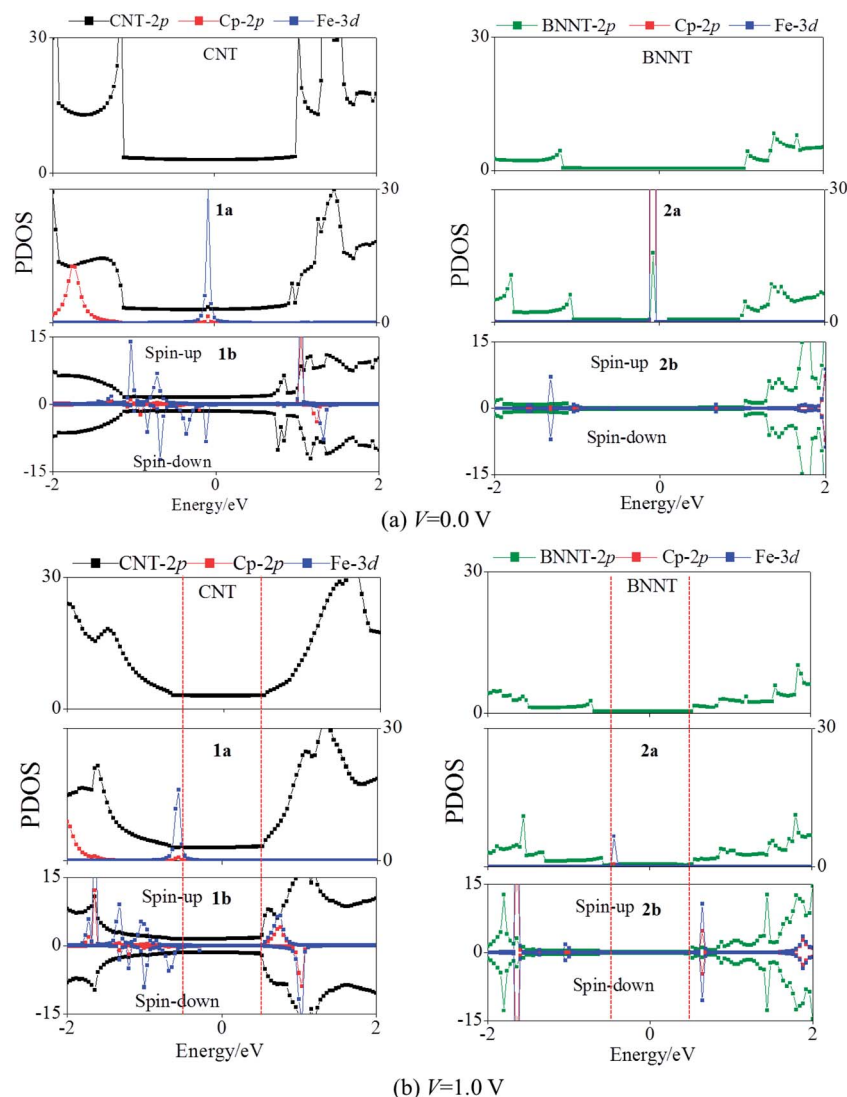


Fig. 7 Computed projected density of states (PDOS) of **1a**, **1b**, **2a**, and **2b** two-probe devices at biases of (a) 0.0 and (b) 1.0 V. For comparison, PDOS of the pure CNT and BNNT two-probe devices are also given. The vertical red lines in (b) refer to the bias window.

from the contour plot of the electrostatic potential along the tube (Fig. 9).

CNT/1b/CNT. At $V = 0.0$ V bias and in the spin-up state of the scatter region **1b** both the HOMO and LUMO stem from the CNT π states (Fig. 6(e)), while in the spin-down state, the HOMO stems from the Fe d orbital and the LUMO originates from the CNT π state (Fig. 6(g)). This is consistent with the electronic structure of the infinite **1b** system. Importantly, under 1.0 V bias, both the HOMO and LUMO, either in the spin-up state or in the spin-down state, are dominated by the CNT π states (Fig. 6f and h). Clearly, no Fe d PDOS appear within the bias window and the bias window exhibits only the CNT π PDOS at the 1.0 V bias (left panel in Fig. 7(b)). Therefore, the conductivity of **1b** is contributed by the CNT π channel in the two-probe device. This is why **1b** has the same I - V curve as pure CNTs and **1a**. Moreover, the spin-up and spin-down states give the same HOMO-LUMO gap of ~ 0.53

eV and the same transition spectrum at E_f and at 1.0 V bias voltage (left panel in Fig. 8(b)), indicating that the conductivity of **1b** is independent of the spin state, consistent with the I - V curves shown in Fig. 5(a). Distributions of electrostatic potential in Fig. 9 also suggest that the core Fe_2Cp_3 does not participate in the electron transport.

CNT/2a/CNT. For pure BNNTs, the MPSH states shown in Fig. 10(a) indicate that the HOMO and LUMO are located at two ends of the finite-size BNNT at $V = 0.0$ V bias. At $V = 1.0$ V bias, the HOMO is located on the right side of the BNNT while the LUMO is located on the left side (Fig. 10(b)). With $\text{Fe}(\text{Cp})_2$ encapsulation, the Fe d state becomes the HOMO and the BNNT π state is still the LUMO at 0.0 V bias (Fig. 10(c)), consistent with the computed electronic structure of the infinite **2a** system. Note that the BNNT valence π state is upshifted and turns into the HOMO at 1.0 V bias (Fig. 10(d)), indicating that the external bias can induce orbital rearrangement. Consequently, the



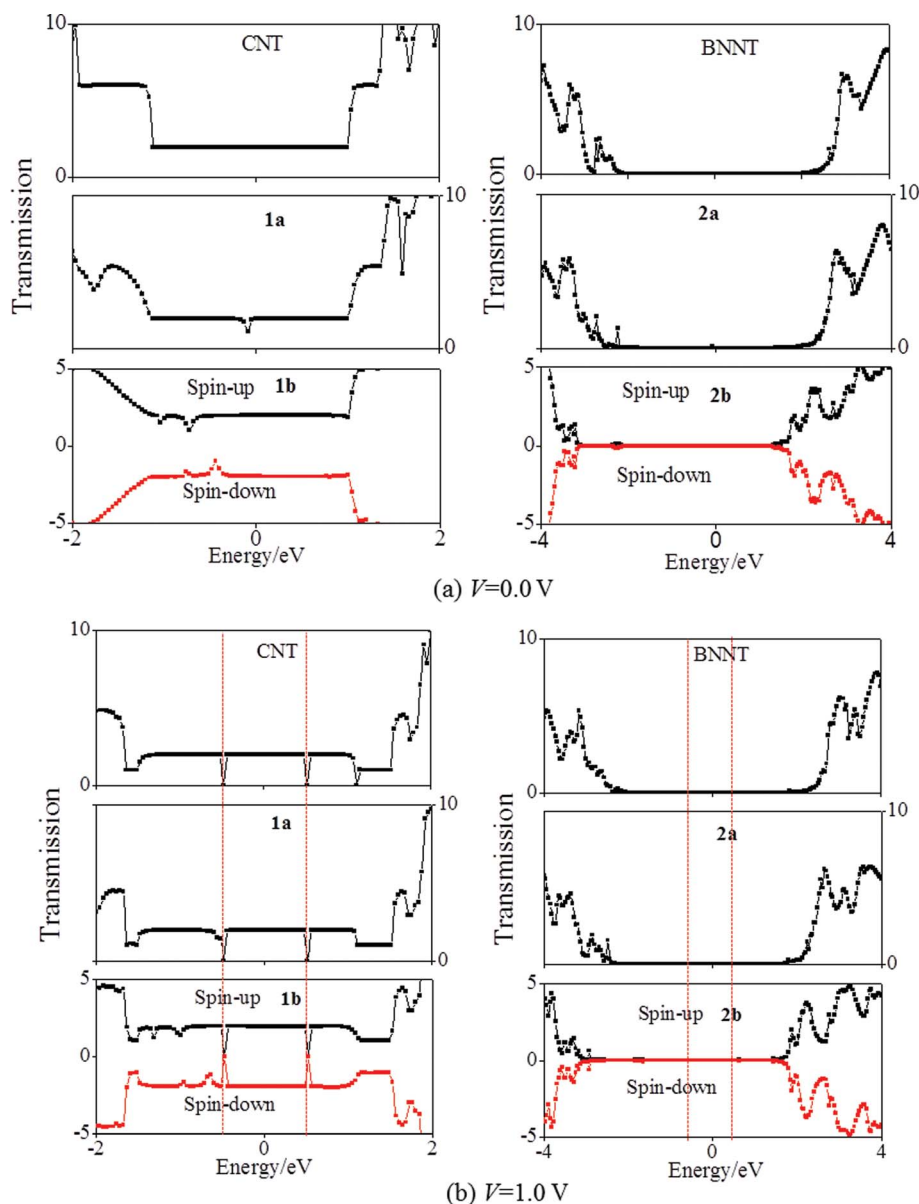


Fig. 8 Transmission spectra (TS) of **1a**, **1b**, **2a**, and **2b** two-probe devices. For comparison, transmission spectra of pure CNT and BNNT two-probe devices are also given. The vertical red lines in (b) refer to the bias window.

HOMO and LUMO of CNT/**2a**/CNT are both controlled by the BNNT π state. Meanwhile, the occupied Fe d PDOS is relocated to the lower energy region at 1.0 V bias but still lies within the bias window (Fig. 7). The TS distributions of **2a** in Fig. 8 clearly show that **2a** is an insulator. Evidently, the MPSH states indicate that the HOMO of CNT/**2a**/CNT becomes more delocalized than the pure BNNT (Fig. 10(b) vs. 10(d)), thereby becoming a more effective transport channel. The electrostatic potential distributions shown in Fig. 9 indicate that the core $\text{Fe}(\text{Cp})_2$ can offer a transport path at 1.0 V bias. In fact, by comparing the electrostatic potential with that of other nanopeapods, the efficiency of electron transport through the core $\text{Fe}(\text{Cp})_2$ in CNT/**2a**/CNT is the highest at 1.0 V bias. Therefore, the conductivity is improved by encapsulating $\text{Fe}(\text{Cp})_2$ into BNNTs, as demonstrated by the I - V curve in Fig. 5(b).

CNT/2b/CNT. Note also that regardless of the bias voltage, no obvious difference in the energy spectra, the PDOS, and the TS between the spin-up state and the spin-down state when the unit of $\text{Fe}_2(\text{Cp})_3$ @BNNT is sandwiched between two CNT electrodes. This is clearly different from the case of the infinite systems which show distinctly different band structures between the spin-up state and the spin-down state. Confirmation of **2b** can be made through the TS distributions in Fig. 8. At 0.0 V bias, the HOMO is controlled by the BNNT π state and the LUMO is dominated by a coupling state of $\text{Fe}_2(\text{Cp})_3$ -BNNT (Fig. 10(e) and (g)). At 1.0 bias, the HOMO and LUMO both stem from the BNNT π state (Fig. 10(f) and (h)), suggesting that the orbital rearrangement occurs under external bias. Through comparison, one can find that the



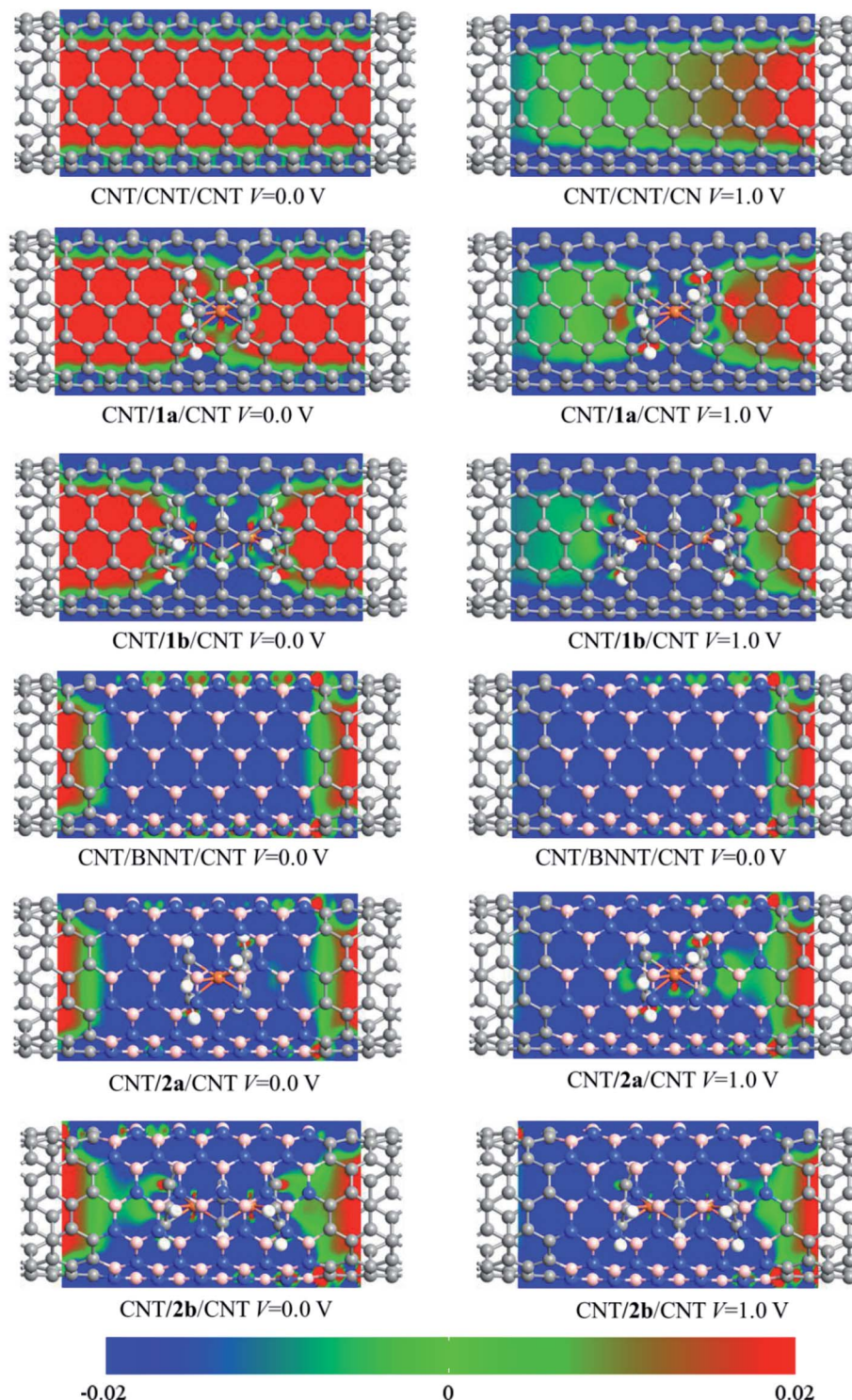
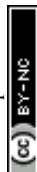


Fig. 9 Computed contour plot of potential distribution for 1a, 1b, 2a, and 2b at 0.0 and 1.0 V. For comparison, potential distribution of the pure CNT and BNNT are also given. Blue (red) represents low (high) potential.

π -delocalization of the HOMO and LUMO is shrunk if an external bias is applied. At 0.0 V bias, the HOMO and LUMO states are located at both ends of the system, while at 1.0 V bias, they are off just to one side of the system. No $\text{Fe}_2(\text{Cp})_3$ PDOS appears within the bias window (Fig. 7). The potential

distribution in Fig. 9 also indicates that the core Fe_2Cp_3 acts as a barrier in the electron transport. Therefore, the conductivity of CNT/2b/CNT should be weaker than the pure BNNT and 2a.



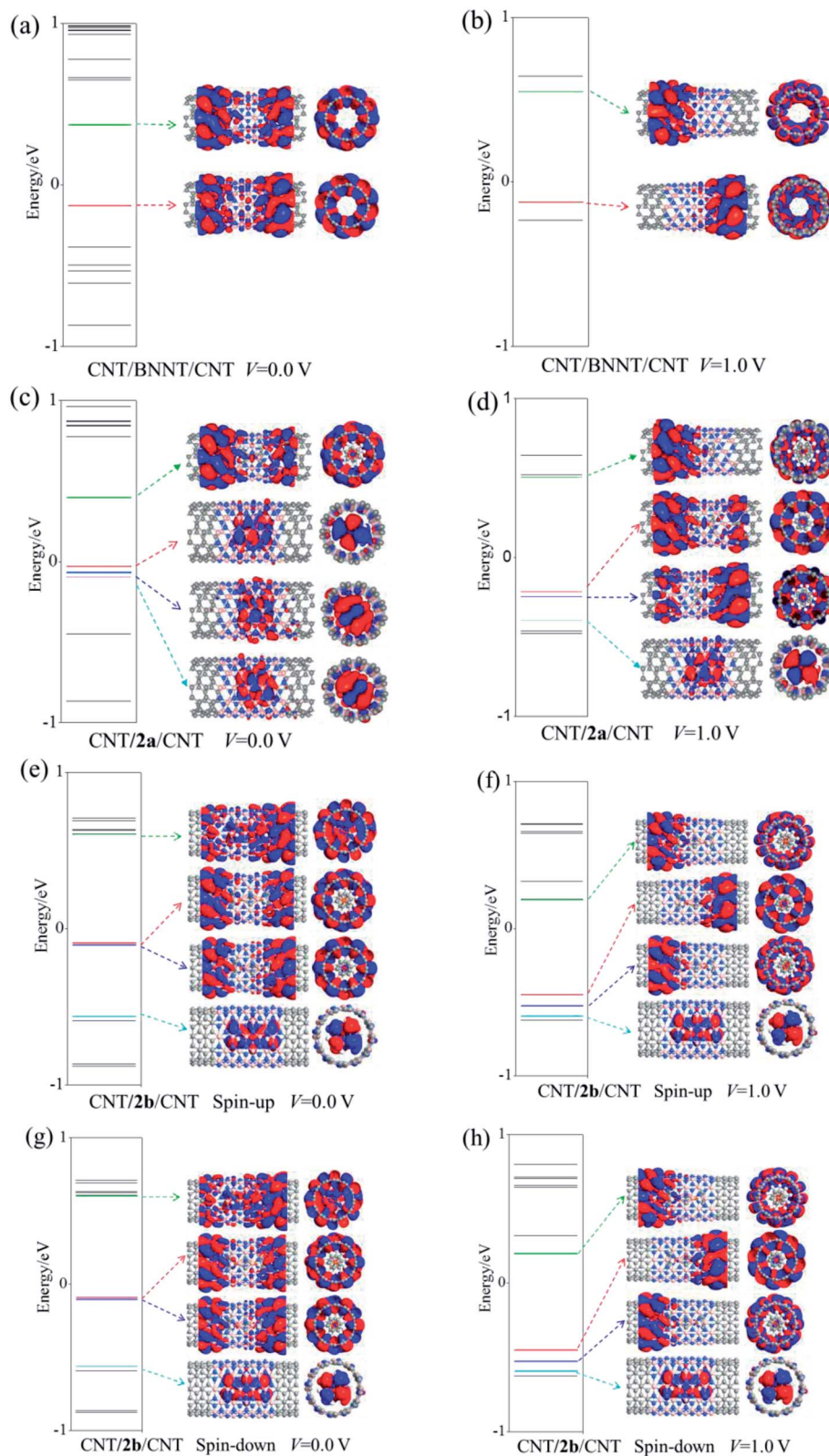


Fig. 10 Computed energy spectra (left panels) of (a and b) the pure BNNT; (c and d) **2a**; and (e–h) **2b** two-probe devices and the MPSH states (middle and right panels) corresponding to the energy levels (highlighted in color lines) near E_f . The iso-surface value is 0.05 ($e \text{ \AA}^{-3}$).



4 Conclusions

We have investigated the electronic and transport properties of a novel form of $\text{Fe}(\text{Cp})_2@ \text{CNT}$, $\text{Fe}_2(\text{Cp})_3@ \text{CNT}$, $\text{Fe}(\text{Cp})_2@ \text{BNNT}$, and $\text{Fe}_2(\text{Cp})_3@ \text{BNNT}$ by means of DFT and NEGF methods. We have found that endohedral encapsulation of $\text{Fe}(\text{Cp})_2/\text{Fe}_2(\text{Cp})_3$ into CNTs (6, 6) or BNNTs (6, 6) is energetically favorable. In $\text{Fe}_2(\text{Cp})_3@ \text{CNT}$ and $\text{Fe}_2(\text{Cp})_3@ \text{BNNT}$ nanopeapods the spin of Fe is coupled ferromagnetically. Conductivities of $\text{Fe}_2(\text{Cp})_3@ \text{CNT}$ and $\text{Fe}_2(\text{Cp})_3@ \text{BNNT}$ nanopeapods appear to be independent of the spin state within the bias range of -1.0 to 1.0 V, even though the band structures and PDOS distributions of the spin-up state and the spin-down state of the infinite structures are different. Characteristics of computed electric conductivities based on the two-probe devices are qualitatively consistent with the computed electronic structures of the corresponding infinite nanopeapod systems.

Encapsulating either $\text{Fe}(\text{Cp})_2$ or $\text{Fe}_2(\text{Cp})_3$ into the CNT has little effect on the conductivity owing to the strong metallic character of the CNT sheath. Encapsulating $\text{Fe}(\text{Cp})_2$ into the BNNT however can notably enhance electron conduction due to added electron hopping from the core $\text{Fe}(\text{Cp})_2$ to the sheath BNNT. Encapsulating $\text{Fe}_2(\text{Cp})_3$ into the BNNT does not assist electron conduction due to the trap effect. Hence, the electronic and transport properties of the BNNT nanopeapods are very sensitive to the length of guest molecules while those of the CNT nanopeapods are insensitive to the length of guest molecules. These properties of BNNT and CNT nanopeapods can be exploited for nanoelectronic and sensor applications.

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