

Self-assembly of chiral tubules†

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The efficient and controlled assembly of complex structures from macromolecular building blocks is a critical open question in both biological systems and nanoscience. Using molecular dynamics simulations we study the self-assembly of tubular structures from model macromolecular monomers with multiple binding sites on their surfaces [Cheng *et al.*, *Soft Matter*, 2012, 8, 5666–5678]. In this work we add chirality to the model monomer and a lock-and-key interaction. The self-assembly of free monomers into tubules yields a pitch value that often does not match the chirality of the monomer (including achiral monomers). We show that this mismatch occurs because of a twist deformation that brings the lateral interaction sites into alignment when the tubule pitch differs from the monomer chirality. The energy cost for this deformation is small as the energy distributions substantially overlap for small differences in the pitch and chirality. In order to control the tubule pitch by preventing the twist deformation, the interaction between the vertical surfaces must be increased without resulting in kinetically trapped structures. For this purpose, we employ lock-and-key interactions and obtain good control of the self-assembled tubule pitch. These results explain some fundamental features of microtubules. The vertical interaction strength is larger than the lateral in microtubules because this yields a controlled assembly of tubules with the proper pitch. We also generally find that the control of the assembly into tubules is difficult, which explains the wide range of pitch values and protofilament numbers observed in microtubule assembly.

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1. Introduction

The self-assembly of macromolecular building blocks into structures of well-defined shapes and sizes is a fundamental challenge in nanoscience.¹ Part of the promise of nanoscience is the development of sophisticated supramolecular assemblies that possess multifunctional properties and behavior beyond the capabilities of simpler molecules.² Biology is full of examples of such super-structures. In particular, microtubules are biopolymers that possess features distinct from standard synthetic polymers because the monomeric building block is a complex protein called tubulin.³ A major feature of microtubules not available in synthetic systems is that they are the track upon which kinesin/dynein motor proteins walk. Fast depolymerization of microtubules is an essential part of their biological function and quite distinct from the typical behavior of synthetic polymers. These special functions and properties originate from the structural arrangement of tubulin in microtubules, which produces very stiff, chiral tubules.

Understanding how the interactions between tubulin monomers yield the self-assembly of microtubules and determine their properties is a fundamental open issue. This issue is pertinent to both biology and materials science, as the special properties of microtubules have led them to be used in synthetic systems to create new materials.^{4–7} These studies have stimulated interests in developing supramolecular systems that act as “artificial microtubules”, *i.e.*, possess features similar or analogous to microtubules. A far-reaching issue inherent to nanoscience is the determination of the essential features of a monomer that self-assembles into a tubular structure mimicking microtubules as well as possessing other features of microtubules.

Microtubules are both a motivation and an inspiration for our work. That is, we want to understand both the assembly of tubulin into microtubules and the more general principles of the assembly of artificial tubular structures. The large amount of data on microtubules helps define our models and provides something to compare to our simulation results. The monomer of microtubules is a dimer of two proteins, α and β -tubulin. The dimers form protofilaments *via* noncovalent longitudinal bonding, and typically 13 protofilaments bind laterally and form tubules with an outer diameter ~ 25 nm.⁸ Variations in the number of protofilaments and the helical pitch of microtubules assembled from free tubulin *in vitro* are significant.^{9,10} The structural variations have also been observed to occur to a smaller extent for *in vivo* microtubules, where other molecules

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help control assembly. These results suggest that the controlled formation of tubules is not simple and understanding the limitations is important for designing synthetic mimics of microtubules.

Tubular structures have been observed in synthetic systems. Recently, a few cases of supramolecular systems have been developed that self-assemble into tubules.^{11–13} Tarabout *et al.* constructed a wedge-shaped nanoparticle from beta-sheet-forming polypeptides including an artificial peptide.¹¹ The nanoparticle consists of two layers composed of beta sheets, which form a bilayer structure with the hydrophilic groups on its top and bottom that sandwich the hydrophobic groups. The top layer is wider than the bottom layer due to larger aromatic residues (including artificial peptides) at the edge of the top layer. They showed that the tubule diameter can be controlled by chemical modifications of an aromatic residue involved in the contact of nanoparticles. Wang *et al.* have found conditions where polypeptide-grafted comblike polymers form tubular structures.¹² Moreover, they found that gold nanoparticles with grafted poly-(L-glutamic acid) can form tubules under certain circumstances.^{12,13} In these systems, the monomer is effectively a single nanoparticle and the nanoparticle–nanoparticle interactions dictate the tubule formation. It is this class of synthetic tubular systems that we are interested in and have developed some new understanding of the role of the interactions between nanoparticles and the control of the self-assembled tubular structures.

There is also a class of small amphiphilic molecules that form tubules typically by initially forming sheets that subsequently roll into tubules.^{14–19} This class of tubular structures is distinct from microtubules (*e.g.*, not as stiff) and is not the focus of this work, although there is some overlap in the underlying assembly phenomena. For example, Shimizu's group developed wedge-shaped amphiphiles by covalently linking hydrophilic groups of different size to the two ends of a hydrophobic spacer.²⁰ They experimentally showed that they can control the inner diameter of multilayered nanotubes by varying the length of the hydrophobic spacer.²¹ The wedge geometry does promote tubular structures, but developing these systems into artificial microtubules appears unlikely.

We have previously developed a coarse-grained model monomer with a wedge shape that can self-assemble into tubular structures with the appropriate binding interactions between monomers.²² The monomer has lateral interactions that promote ring formation and vertical interactions that promote filament formation. We used molecular dynamics (MD) simulations to study the self-assembly process and obtained a diagram of the self-assembled structures as a function of lateral and vertical interaction strengths. Since our model monomers can self-assemble into tubular structures and the resulting tubules exhibit similar structural polymorphism as microtubules, we have a model that can be used to systematically examine the physical origin of structural variations and explore ways to control the structure. Furthermore, the strength of lateral and vertical bonds between tubulin monomers clearly plays important roles in determining the structure of microtubules, but their effects are hard to probe experimentally since natural evolution only leads to one particular set of

interaction strengths between tubulin.²³ Directly measuring the interactions between monomers is typically not possible because experiments usually only determine the net free energy difference that is the sum of many interactions. Our model system offers an opportunity to directly calculate the energetics and the structural dynamics of tubules in more detail than previously attempted.

The focus of this work is the physical origin of the factors controlling tubule assembly. To this end, we have added new features to our model monomer to improve the structural control of the self-assembled tubules based on new understanding of the physical nature of the assembly process. In particular, we focus on controlling the helicity of self-assembled tubules and have added an explicit chirality to the monomer so that the desired tubule helicity can be input from the monomer chirality. Unexpectedly, our earlier simulations produced helical tubules but with achiral monomers.²² We show here that this tubule helicity is a consequence of an underlying rotational symmetry that can occur through a twist deformation of protofilaments. Twist of tubules also occurs with chiral monomers and enables a range of tubule helicity about the desired value to occur, *i.e.*, the monomer chirality. In other words, mismatch occurs between the chirality of building blocks and the pitch of assembled tubules. We incorporate a lock-and-key binding mechanism for the vertical interactions between monomers to more precisely constrain the tubule helicity. Our studies show that the lock-and-key binding combined with monomers with built-in chirality can be used to achieve structural control in the self-assembly of tubules, and therefore provide a guidance on the design of building blocks that will efficiently self-assemble into controlled tubular structures.

The remainder of this paper is organized as follows. The simulation methods are briefly described in the next section. In the Results and discussion section, we first examine the energetics of the achiral system and explain the physical origin of the helical tubules that assemble from the achiral wedges. We then include explicit chirality in the monomer design and present the assembly and energetics for the chiral systems. Finally, we discuss the results of the systems that incorporate both chirality and a lock-and-key interaction for the vertical binding. These systems provide the best control of the assembly of tubular structures. Conclusions are included in the last section.

2. Simulation methods

The wedge-shaped building blocks are shown in Fig. 1. Each monomer is treated as a rigid body in simulations reported here. The core of the monomer is a wedge deformed from a cube composed of 27 particles (gray spheres in Fig. 1) with an undeformed lattice constant 1σ , where σ is the particle diameter. The deformation is imposed in such a way that 13 wedges will fit to form a closed ring. The interaction between these core sites on two monomers is modeled as a short-range repulsion using the Lennard-Jones 12–6 potential $U_{LJ}(r) = 4\epsilon [(\sigma/r)^{12} - (\sigma/r)^6]$, where ϵ is the energy scale. This potential is cut off and shifted to 0 at $r_c = 1.0\sigma$. The short-range repulsion models the excluded volume interaction between monomers.



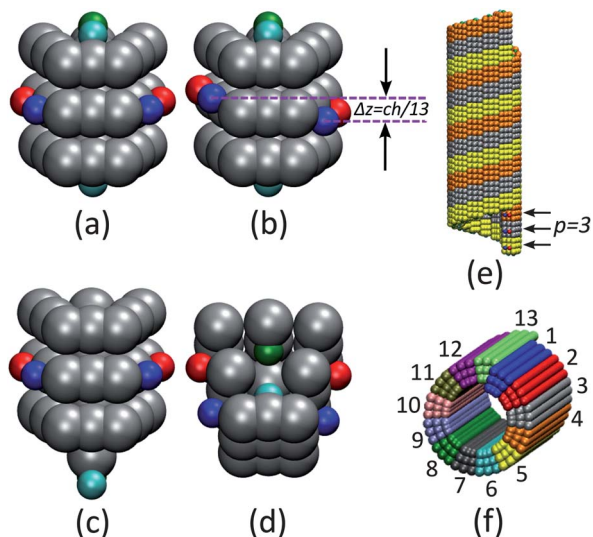


Fig. 1 (a) An achiral wedge monomer M_0 designed for nonhelical tubules (*i.e.*, pitch $p = 0$). (b) A chiral wedge monomer M_2 with chirality $c = 2$ (*i.e.*, designed for tubules with $p = 2$). The chirality c determines *via* $\Delta z = ch/13$ the displacement Δz between the binding sites on the left and right sides of the wedge, where $h = 3\sigma$ is the wedge height. (c) An achiral wedge monomer with a lock-and-key configuration for the vertical binding, M_0^{LK} ; the vertical binding sites stick out at the bottom surface and are indented at the top surface. (d) A chiral wedge monomer with $c = 2$ and with a lock-and-key configuration for the vertical binding, M_2^{LK} ; the top view shows that the vertical binding sites are buried below the top surface. (e) A 13_3 tubule with $N = 13$ protofilaments and pitch $p = 3$ (each helix is colored differently). (f) The same tubule as in (e) but with a top view showing 13 protofilaments (each protofilament is colored differently); for clarity the binding sites are not shown in (f).

The temperature T is set as $1.0\epsilon/k_B$ (k_B is the Boltzmann constant) in our simulations. We will use $k_B T$ as the energy unit. Attractions between monomers occur through 8 binding sites (colored spheres in Fig. 1) placed on the lateral and vertical faces of the wedge. A binding site on the left (top) face of a wedge only interacts with the binding site in the same color and on the right (bottom) face of another wedge and *vice versa*. The bonding interaction is modeled as a soft potential $U_B(r) = -A[1 + \cos(\pi r/r_a)]$, where A is the binding strength and r_a the interaction range. The well depth of this potential is $2A$ at $r = 0$. We use A_L and A_V (in the unit of $k_B T$) to designate the strength of the lateral and vertical binding interactions, respectively. We use this potential form because it smoothly goes to zero at $r = r_a$. In our previous work and this paper, we fix $r_a = 1.0\sigma$. The potential form allows for variation of r_a , but we leave that for future work. Since $U_B(r)$ is isotropic, a minimum of 2 binding sites on each face is needed to break the rotational symmetry when two monomers bind, which is crucial to ensure that a pair of bonded monomers have the proper orientation.

In the original design, the lateral binding sites are placed at 0.5σ outside the lateral face and in the middle plane of the core particles, as shown in Fig. 1(a). The vertical binding sites are similarly placed with respect to the top or bottom face. Such a building block has a mirror symmetry and its achiral geometry

is designed for nonhelical tubules. This monomer is designated as M_0 . In this work, we extend the wedge model to treat chiral monomers. Chirality is introduced by shifting the lateral binding sites oppositely on the left and right side of the wedge, respectively. In general, a chiral monomer will be designated as M_c , where c designates the chirality of the monomer. For a M_c monomer, the displacement between the lateral binding sites is $\Delta z \equiv ch/13$, where $h = 3\sigma$ is the wedge height. The M_2 monomer is shown in Fig. 1(b). We treat cases where the M_c monomers are designed to assemble into tubules with pitch $p = c$. To label tubular structures, we follow the literature and use N_p to denote tubules with N protofilaments and pitch p (counted in the unit of building blocks and also called *helix start number* in the literature on microtubules).¹⁰ An illustration of p and N is given in Fig. 1(e) and (f).

One surprise found in our previous simulations using the achiral M_0 is the preferential assembly of helical tubules with pitch 1 or even 2.²² New results here show that tubules with pitch mismatching the monomer chirality still assemble even for chiral monomers. We found that one factor that can help suppress the mismatch and better control the tubule pitch is to make vertical binding stronger. Since simply increasing A_V leads to kinetically trapped clusters of wedges, we instead explore a lock-and-key mechanism for the vertical binding interactions, which is introduced by modifying the location of the vertical binding sites. These sites together with the central line of core particles are displaced vertically by 0.75σ , so that the top pair of binding sites are buried below the top surface of the wedge by 0.25σ , while the bottom pair stick out of the bottom surface by 1.25σ . The lock-and-key modification and chirality are combined to make building blocks labeled M_c^{LK} , where the superscript LK stands for lock-and-key. M_0^{LK} and M_2^{LK} are shown in Fig. 1(c) and (d), respectively.

All simulations were performed with the LAMMPS simulation package. The equations of motion were integrated using a velocity-Verlet algorithm with a time step $\delta t = 0.005\tau$, where $\tau \equiv \sigma(m/\epsilon)^{1/2}$ is the unit of time and m the mass of one site. The simulations studying the self-assembly of various monomers involved 1000 wedges and were run for $4 \times 10^6\tau$ to $8 \times 10^6\tau$. The initial state was a low-density gas of monomers uniformly distributed in the simulation box. The temperature of the systems was kept at $1.0\epsilon/k_B T$ with a Langevin thermostat of damping rate $1.0\tau^{-1}$.

3. Results and discussion

3.1. Helicity of tubules assembled from achiral wedges

The first important issue to be addressed is why achiral monomers form helical tubules. To this end, we built 13_p tubules from M_0 monomers with p ranging from 0 to 3. The tubules start in a state with straight protofilaments. The energy distribution densities per monomer, $D(E)$, calculated for each p from MD simulations after the tubules reach equilibrium are shown in Fig. 2(a). As expected, the 13_0 tubule has the lowest mean energy. However, the energy distribution of the 13_1 tubule has a large overlap with that of the 13_0 tubule. Even the energy distribution of the 13_2 tubule overlaps with that of the



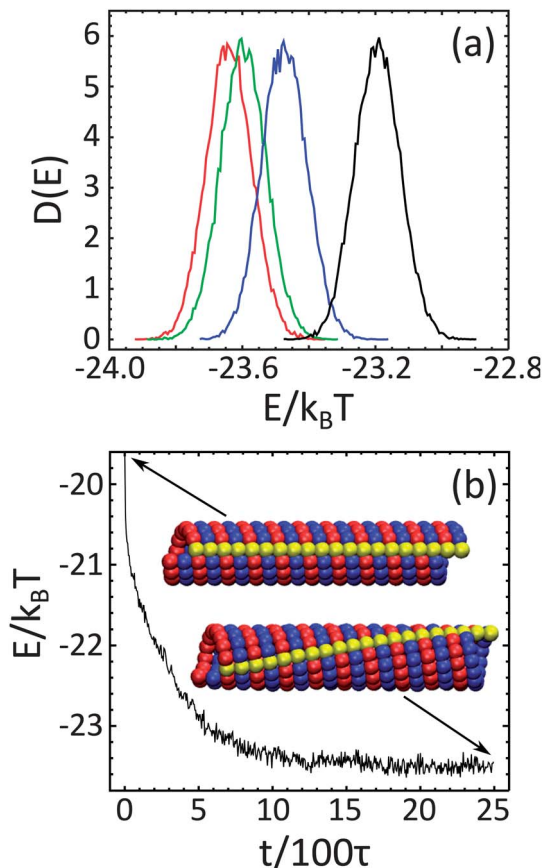


Fig. 2 (a) The probability density of energy distribution per monomer, $D(E)$, for equilibrated 13_p tubules with $p = 0$ (red), 1 (green), 2 (blue), and 3 (black), from left to right. (b) A 13_2 tubule starting with straight protofilaments evolves into a lower-energy state with twisted protofilaments. In the images each wedge making up the tubule is shown as a single sphere. The helices formed *via* the lateral binding are shown in red and blue, respectively, and one protofilament is colored yellow to emphasize the twist transformation. Both (a) and (b) are for $A_L = 4.2$ and $A_V = 2.6$.

13_0 tubule. Being so close energetically explains the frequent formation of 13_1 and 13_2 tubules by the achiral M_0 monomers in the assembly simulations starting with free monomers (*i.e.*, the initial condition is a “gas” phase of wedge monomers). The overlap becomes negligible as p increases to 3, which explains the lack of 13_3 tubules in the self-assembly products. While the energy difference per monomer at the peaks of $D(E)$ for 13_0 and 13_3 is only about $0.4k_B T$, the difference between the initial tubules formed in the self-assembly simulations is much larger since many monomers are involved in the nucleation and the total energy difference depends on the initial tubule size. A single turn of 13 monomers puts the total energy difference above $5k_B T$. In some cases ($A_V > A_L$) multiple partial turns form before the tubule state occurs, and the energy difference will be consequently much larger than $5k_B T$.

A structural transformation must occur for the $p = 1$ or 2 tubules to have energies that overlap with $p = 0$ because helical tubules (*i.e.*, $p > 0$) with straight protofilaments have the neighboring lateral attractive sites displaced resulting in higher

energies. To probe the role of twist deformation, we examined the energy and geometry for the 13_2 tubule. Starting from straight protofilaments as shown in the top image of Fig. 2(b), this tubule quickly transforms into a steady state with twisted protofilaments as shown in the bottom image of Fig. 2(b). During the transformation, the energy per monomer decreases with time by about $4k_B T$ as shown in Fig. 2(b). The energy decreases as protofilaments get twisted because the twist deformation brings the lateral interaction sites into alignment. The protofilaments in the nonhelical 13_0 tubule do remain straight as expected.

Fig. 3 illustrates the twist transformation at the monomer level with the full geometry and local rotation of wedges visualized, showing how twisted protofilaments can help lower the energy of a helical tubule made out of M_0 monomers. As expected for straight protofilaments in a 13_2 tubule (Fig. 3(a) dotted red line), the lateral binding is clearly not maximized because of the displacement between the lateral attractive sites. As shown by the dotted red lines in Fig. 3(b), in the twist transformation the rotation of monomers aligns the lateral sites bringing their binding strengths close to their maxima. Consequently, the total energy of a helical tubule is reduced, which stabilizes the tubule. Thus the energy of tubules with $p \neq c$ can be close to the preferred $p = c$ case because the twist deformation of the protofilaments substantially alters the energy by making the lateral alignment closer to the ideal configuration.

A more detailed examination of the packing of M_0 wedges in helical tubules reveals the limit of twist deformation. In a twisted protofilament, the wedges must rotate about their vertical axis so that its inner surface always points to the interior of the tubule, which introduces an offset between the vertical binding sites of two consecutive stacking wedges, as shown in Fig. 3(c). For M_0 monomers, this offset is very small in a 13_1 tubule, which makes it energetically close to the 13_0 tubule. However, as the pitch of a helical tubule gets larger, the offset increases as a result of the increasing amount of twist. The offset is clearly visible for a 13_2 tubule as in Fig. 3(c) and even more dramatic for a 13_3 tubule (see Fig. S1 in the ESI†). For this reason, the energy distributions for the 13_2 and 13_3 tubules shift to higher and higher values, and the 13_3 distribution is well separated from the 13_0 one. Only so much twist can occur without resulting in an expensive mismatch between the vertical binding pairs.

The result of multiple pitch values occurring for tubules in thermal equilibrium reveals a limit of treating a microtubule as an elastic tubule within the framework of continuum elasticity, where the twist of the tubule always costs energy and induces a restoring force.^{24–27} In our simulations, the continuum theory breaks down because of the discrete nature of the building blocks. The monomers are not deformed in the twist deformation, and thus there is no internal elastic cost. All the interactions are between the surfaces, and the lateral and vertical sites play distinct roles with the energetic cost occurring primarily at the vertical contacts. While the lowest energy state is the same in either treatment (*i.e.*, the tubule with pitch that matches the monomer chirality has the lowest energy), the



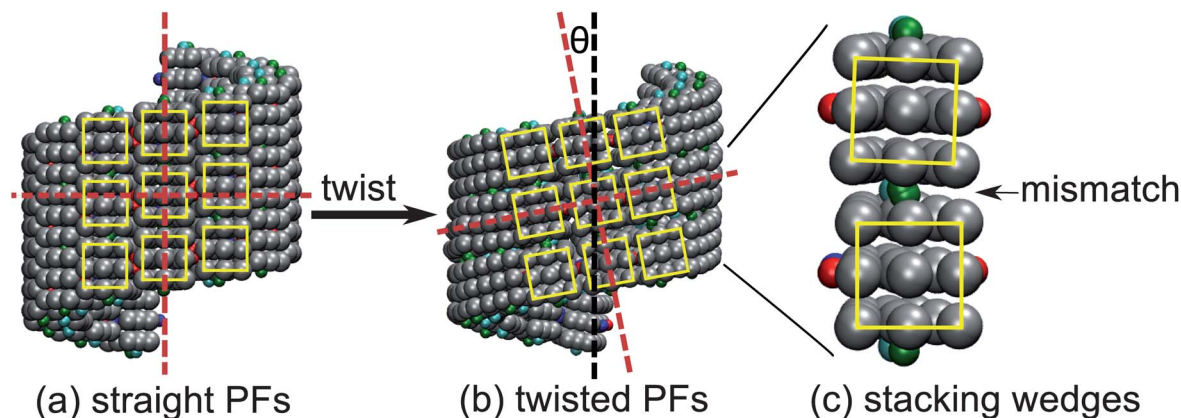


Fig. 3 A 13₂ tubule starting with straight protofilaments (a) undergoes a skew deformation to transform into a tubule with twisted protofilaments (b) at $A_L = 4.2$ and $A_V = 2.6$, which results in a better packing of wedges. For clarity only a short part of the whole tubule is shown. Twist angle θ is the angle between protofilaments and the central axis of the tubule. (c) The stacking of two consecutive wedges in a twisted protofilament involves an offset between the binding sites at vertical interfaces due to the rotation about the central axis.

simulations with discrete building blocks reveal multiple states in equilibrium and the role of the surface interactions between the monomers.

The twist deformation of protofilaments can be quantified by a twist angle θ , which is the angle between the protofilaments and the central axis of the tubule (Fig. 3(b)). For a N_p tubule, θ depends on both N , p , and c , and their relation can be easily derived through geometrical considerations based on the packing of anisotropic objects on the curved surface of a tubule, called the lattice accommodation model in the literature.¹⁰ In general, for a N_p tubule built from M_c monomers, the twist angle θ is given by

$$\tan \theta = \frac{h}{w} \left(\frac{p}{N} - \frac{c}{N_0} \right), \quad (1)$$

where h and w are the height and width of the wedge, respectively, and $N_0 = 13$ is the designed number of protofilaments in an ideal tubule. Values of θ for tubules assembled from M_0 are shown in Fig. 4. In this case, eqn (1) indicates that θ is always 0 for tubules with $p = 0$, no matter how many protofilaments the tubule contains. The results for N_0 tubules in Fig. 4 is consistent with this prediction. For N_p tubules with $p \neq 0$, θ depends on N via eqn (1), as confirmed by the corresponding simulation results in Fig. 4. We have also studied the effects on θ of both A_L and A_V . As expected, θ is generally insensitive to either A_L or A_V because θ is mainly determined by the geometric features (*i.e.*, width, height, and chirality) of the building blocks as expressed in eqn (1). However, one thing to notice is that results in Fig. 4 are obtained with tubules starting with pre-determined pitch and protofilaments that are appropriately twisted according to eqn (1). In this case the pitch does not change during the simulation and the tubule only fluctuates around the steady configuration, which is close to the starting one but generally only metastable. The situation becomes much more complicated if we use tubules starting with straight protofilaments, where the relative strength of A_V and A_L plays an important role in affecting the stability of the tubules. More details are given in the ESI (see Fig. S2–S4†).

The above results show that the vertical interaction is important in limiting the range of tubule helicity since it regulates the twist of protofilaments. Yet, we have found that increasing A_V to increase the energy cost of twist is insufficient to control the helicity of tubules. Energy distributions of pre-built tubules with various pitch values only show slightly less overlap even for $A_V \gg A_L$. For example, the energy distributions for the 13₀ and 13₁ tubule shown in Fig. 5(a) for $A_V = 6.3$ look very similar to that in Fig. 2 for $A_V = 2.6$. For the 13₂ tubule the overlap with the energy distribution of the 13₀ tubule is clearly reduced, but some overlap still exists. In general, the reduction in the energy overlap at large A_V is not sufficient to change the range of tubule helicity; tubules with $p > 0$ will still self-assemble for M_0 monomers even for very large A_V . Moreover, self-assembly simulations of free monomers with large A_V actually

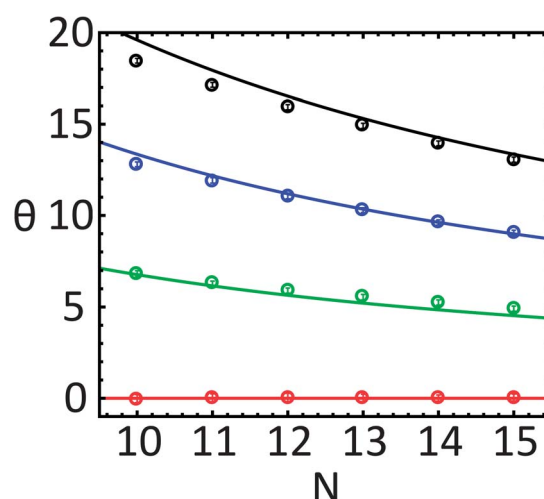


Fig. 4 Twist angle θ of tubules formed by M_0 monomers and with various number (N) of protofilaments and pitch values: $p = 0$ (red), $p = 1$ (green), $p = 2$ (blue), and $p = 3$ (black) at $A_L = 3.0$ and $A_V = 4.2$. Symbols are simulation results with error bars comparable or smaller than the symbol size. Lines represent the corresponding predictions of eqn (1) with $h = 3\sigma$ and $w = 2.53\sigma$.



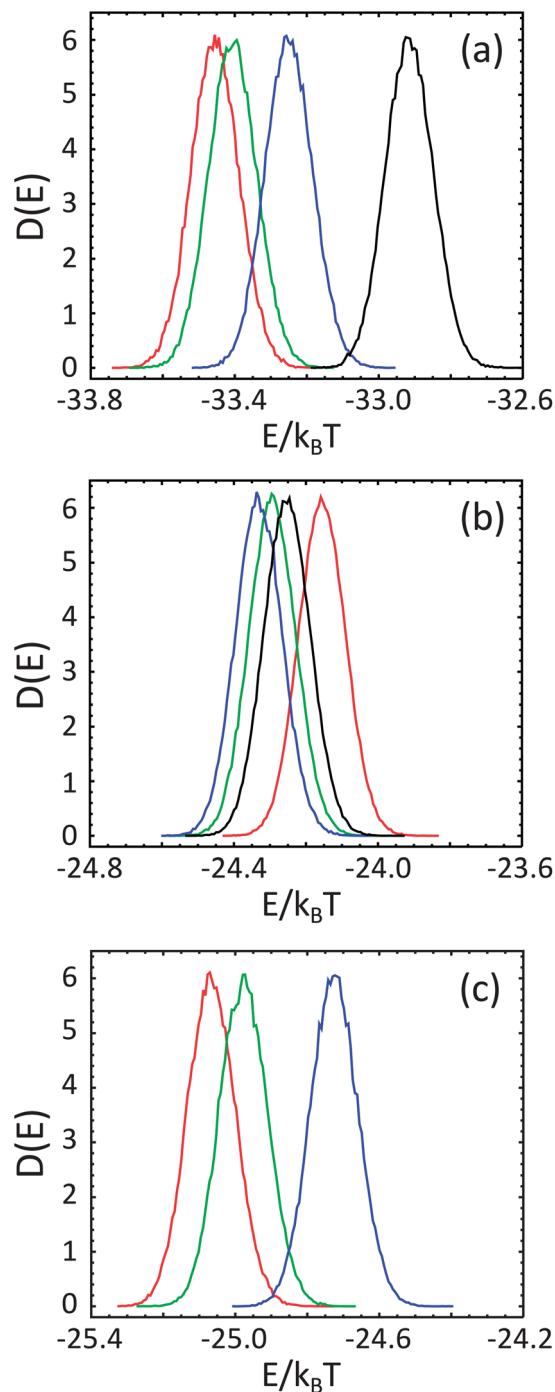


Fig. 5 The probability density of energy distribution per monomer, $D(E)$, for various 13_p tubules: $p = 0$ (red), 1 (green), 2 (blue), and 3 (black). (a) M_0 at $A_L = 3.0$ and $A_V = 6.3$; (b) M_2 at $A_L = 3.0$ and $A_V = 3.9$; and (c) M_0^{LK} at $A_L = 3.0$ and $A_V = 6.3$, where the 13_3 tubule is unstable and its $D(E)$ not calculated.

yield kinetically trapped clusters and other defected structures.²² Thus, these results imply that to control the tubule helicity additional features (*e.g.*, interactions) will have to be added to the monomer.

While assembly kinetics is not the focus of this paper, it plays an important role in the self-assembly of tubules and a

few important points need to be made. The self-assembly simulations start with free monomers uniformly distributed with a low density in a box. When $A_L > A_V$, M_0 monomers tend to first form rings or helical strands, which then grow into nonhelical or helical tubules, respectively.²² On the contrary, if $A_V > A_L$, then M_0 monomers tend to form protofilaments first, which then form curved sheets (see examples in Fig. 6(f)) that eventually close up into tubules. This process is similar to the proposed self-assembly pathway of tubulin into microtubules.²⁸ The final structure of the tubule is strongly affected by the closure event between the two edges of a curved sheet. For M_0 monomers, if A_V is only slightly stronger than A_L , then the tubule can still become helical because the sheet is flexible and thermal fluctuations can introduce an offset between the two edges of the sheet when they meet and close up. If the curved sheets were stiff enough to suppress the effects of thermal fluctuations, then the closure event would be more controlled by the intrinsic chirality of the monomer. However, for the range of interaction strengths at which wedges do self-assemble into tubules, the curved sheets are typically not stiff enough to make thermal fluctuations negligible.

The flexibility of the curved sheets impacts the number of protofilaments of a tubule as well. Especially at $A_V > A_L$, assembled tubules tend to have protofilament number fewer than the designed value 13. Besides the twisting fluctuation in the sheets, there are circumferential fluctuations that tend to bring the opposite sides of the sheet into contact and closure when the number of protofilaments is only 11 or 12, even though the monomer width is chosen to allow 13 wedges to fit from purely geometric considerations. After the closure, it is virtually impossible for other wedges or protofilaments to squeeze into the lattice to make 13-protofilament tubules. In other words, tubules are kinetically trapped in states with protofilaments fewer than designed. It remains an interesting open question that how the design of wedge monomers can be tweaked to address the issue of kinetic trapping and to promote the formation of tubules containing 13 protofilaments. One obvious possibility is to squeeze the width of wedges so that sheets containing fewer than 13 protofilaments are unlikely to close. This direction will be explored in future work.

The previous discussion on the effects of A_L and A_V on the tubule twist, tubule energy distributions, and assembly kinetics provides the basis to understand the self-assembly of systems starting with free monomers, *i.e.*, the initial state is a gas phase. Results on various systems containing 1000 monomers are shown in Fig. 6. Here we just discuss parts (a) and (b), which involve the M_0 monomer. In Fig. 6(a) where $A_L = 4.2 > A_V = 2.6$, all assembled tubules are helical with pitch 1 or 2. We have performed more than one simulation for these parameters and nonhelical tubules ($p = 0$) do form in some cases, but this result is indicative of the overall finding that helical tubules are more common than nonhelical even though the monomer has no chirality, as long as $A_L > A_V$. When the self-assembly is induced at $A_V = 3.9 > A_L = 3.0$ (Fig. 6(b)), then the pitch is 0 or 1, closer to the monomer chirality. In both cases N is 11 or 12, which is a result of the closure dynamics discussed above.



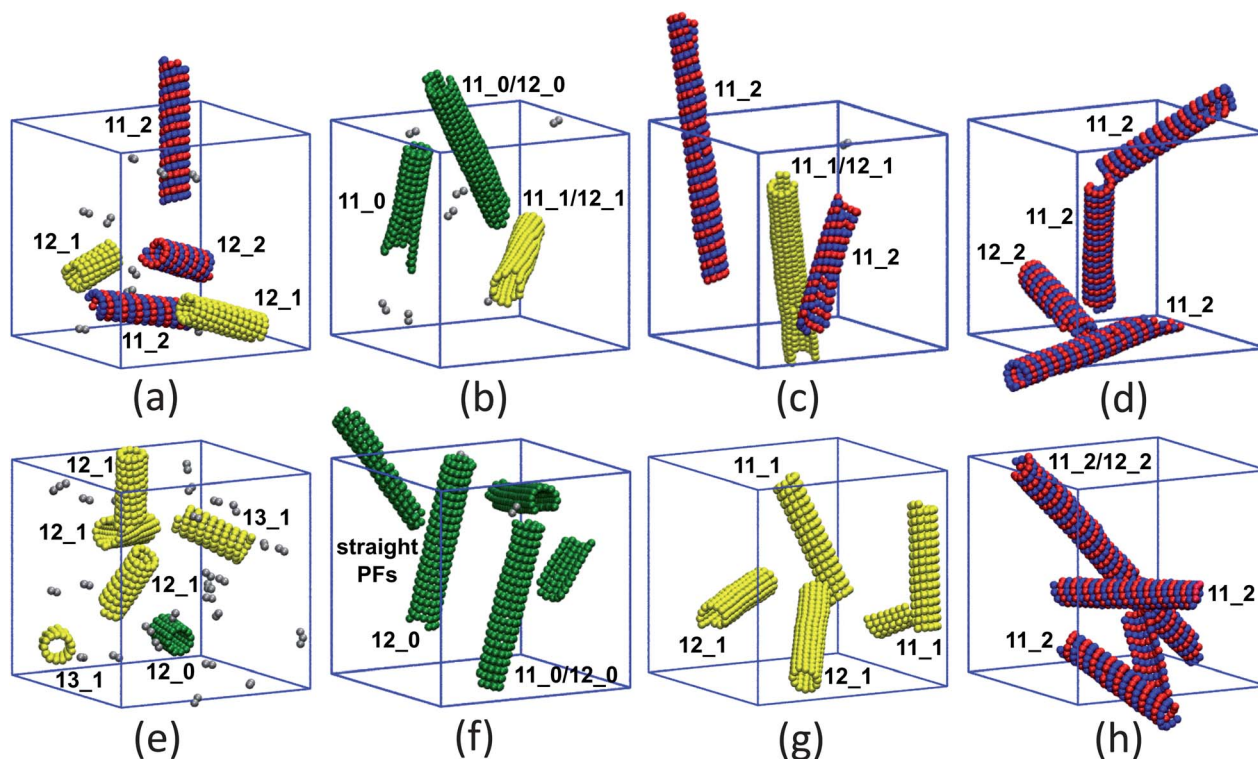


Fig. 6 The self-assembly of tubules with various monomers: (a) M_0 at $A_L = 4.2$ and $A_V = 2.6$; (b) M_0 at $A_L = 3.0$ and $A_V = 3.9$; (c) M_1 at $A_L = 3.0$ and $A_V = 3.9$; (d) M_2 at $A_L = 3.0$ and $A_V = 3.9$; (e) M_0^K at $A_L = 4.4$ and $A_V = 4.2$; (f) M_0^K at $A_L = 3.0$ and $A_V = 6.3$; (g) M_1^K at $A_L = 3.6$ and $A_V = 5.4$; (h) M_2^K at $A_L = 3.0$ and $A_V = 6.3$. Each sphere represents a wedge monomer. The color code is as follows. Structures shown in green are either nonhelical tubules (*i.e.*, $p = 0$) or unclosed sheets with straight protofilaments. Tubules with $p = 1$ are shown in yellow. Tubules with $p = 2$ are shown with helices colored in red and blue, respectively. Oligomers are shown in silver and free monomers are not included.

3.2. Self-assembly of chiral wedges

We now discuss results for the self-assembly of chiral monomers. Chirality is important for better modeling microtubules, which have pitch 3. Chiral wedges are produced by introducing an offset along the vertical direction between the two sets of lateral attractive sites on the opposite sides of the wedge. The amount of offset sets the value of chirality. As noted earlier, ideally the M_c monomer would yield a tubule with pitch $p = c$. The M_2 monomer is shown in Fig. 1(b).

We first present the energy distribution of prebuilt tubules with various pitch values. An example is shown in Fig. 5(b) for the M_2 monomer at $A_L = 3.0$ and $A_V = 3.9$. As expected, the $p = 2$ tubule now has the lowest mean energy per monomer. The $p = 1$ and 3 tubules have the next two higher average energies per monomer. Their energy distributions show large overlaps with that of the $p = 2$ tubule. Even the energy distribution of the $p = 0$ tubule has some overlap with the rest. The spread of the four distributions is smaller than for the M_0 case in Fig. 5(a). For the M_2 (generally M_c) monomer, the protofilaments in tubules with pitch $p \neq 2$ (generally $p \neq c$) are twisted, similar to those in the helical tubules made of M_0 monomers (Fig. 2(b)). The twist angle is found to be consistent with the prediction of eqn (1). Thus, while including chirality in the monomer shifts the lowest energy state to the tubules with the preferred pitch (as input from the monomer chirality), the twist deformation of

protofilaments still yields overlapping energy distributions with mismatched pitch values.

Images of the tubules formed during the self-assembly simulations of M_0 , M_1 , and M_2 at $A_V = 3.9 > A_L = 3.0$ are shown in Fig. 6(b)–(d), respectively. As the value of c shifts so does the pitch p of the assembled tubules. The results are similar between the M_0 and M_1 cases. There are tubules with $p = c$ formed, but there are also tubules with other pitch values. Surprisingly, the self-assembly of M_2 monomers shows that all the assembled tubules have pitch 2, which is the same as the built-in chirality of the M_2 monomers. Three independent runs with different initial conditions all produce tubules only with $p = 2$. But our expectation is that this is just a consequence of small samples. For the three cases in Fig. 6(b)–(d), when the interaction strengths are switched to stronger lateral binding over vertical ($A_L = 4.2 > A_V = 2.6$), a wide range of pitch values occurs (see Table 1 in the ESI†). Particularly in the strong A_L limit the tubules with mismatched pitch values seem to dominate, indicating the effects of strong lateral bonds on the nucleation of seeding structures and the assembly pathway. More discussion is included at the end of Section 3.3. Overall, the data indicate that though the pitch of tubules can be varied in a certain range by using chiral monomers, it is difficult to achieve a precise match between the pitch and chirality. The pitch of assembled tubules usually shows a range centered on the chirality of monomers with $|p - c| \leq 2$.



3.3. Self-assembly of wedges with lock-and-key binding

We have found that the twist deformation makes control of the tubule pitch difficult in most cases. While large A_V would limit or even suppress twist, it generally leads to kinetically trapped structures rather than well-defined tubules. An additional feature or modification of the vertical interaction is required in order to prevent twist. To this end, we introduce a lock-and-key mechanism for the vertical binding. The vertical binding sites together with the central line of core sites of the wedge are shifted along the vertical direction by 0.75σ such that on the bottom face the attractive sites stick out, while on the top face the attractive sites become buried below the surrounding core sites. In this way a simple lock-and-key configuration is created. The vertical lock-and-key (LK) modification on M_c monomers is labeled M_c^{LK} . Images of the M_0^{LK} and M_2^{LK} monomer are shown in Fig. 1(c) and (d), respectively.

The energy distributions for prebuilt tubules of M_0^{LK} monomers are shown in Fig. 5(c) at $A_V = 6.3 > A_L = 3.0$. These interaction strengths are just strong enough to induce self-assembly for this lock-and-key monomer as determined by our assembly simulations starting with free monomers. In general, the addition of the lock-and-key requires a larger value of A_V for the monomers to bind because extra energy is needed to insert the key and to compensate for the stronger repulsion between the core sites (see Fig. S5 and relevant discussion in the ESI†). Compared with tubules of monomers without the vertical lock-and-key binding, overlaps between the energy distributions for tubules with various pitch values are reduced in the lock-and-key case. For the M_0^{LK} monomer, while a significant overlap still exists for tubules with $p = 1$ and $p = 0$, the overlap between $p = 2$ and $p = 0$ is almost completely gone and tubules with $p = 3$ are not even stable anymore. Thus, tubules with the lowest energy per monomer (e.g., tubules with $p = 0$ for M_0^{LK} monomers) will be more favored during the self-assembly of lock-and-key monomers. The self-assembly of M_0^{LK} monomers with the same interaction strengths as for Fig. 5(c) is shown in Fig. 6(f). Again, these simulations start with a gas phase of monomers. For M_0^{LK} , all the assembled tubules are nonhelical with pitch 0, matching the chirality of M_0^{LK} . Some clusters are curved sheets that do not close up into tubules yet because of the exhaustion of free monomers. However, in these sheets the protofilaments are straight, and if more monomers were supplied to the system, the tubules would be expected to be nonhelical when they eventually form.

Two examples of the self-assembly of chiral monomers with the lock-and-key geometry are shown in Fig. 6(g) and (h) for M_1^{LK} and M_2^{LK} , respectively. They show that the lock-and-key mechanism produces a match between the pitch of tubules and the chirality of monomers. This match is better than what we expect from the energy distributions, which imply that mismatched cases should occur. We expect that simulations of larger systems that can form many tubules would produce a distribution of pitch values. Nonetheless, as the comparison between Fig. 6(c) and (g) indicates, the addition of the lock-and-key mechanism for the vertical binding substantially shifts the assembled structures toward tubules with pitch matching the

monomer chirality. Results of the self-assembly simulations of M_c^{LK} monomers at various combinations of A_L and A_V are summarized in Table S2 of the ESI.† This table shows that there are some (A_L, A_V) that yield only $p = c$ tubules, but also there are cases with similar (A_L, A_V) that have $p \neq c$, which is what we expect based on the energy distributions. Overall, the results are indicative of an improved control of the tubule helicity and demonstrate the basic concept that strong vertical interactions are required in order to limit the amount of protofilament twist and to control the tubule pitch.

The relative importance of vertical interactions can be seen in the comparison between Fig. 6(e) and (f) for $A_L > A_V$ and $A_V > A_L$, respectively, for the M_0^{LK} monomer. For the case with stronger lateral interactions, while the chirality range is narrower and closer to the monomer chirality with the lock-and-key binding than without (compare Fig. 6(a) and (e)), the most common tubule formed by M_0^{LK} has pitch $p = 1$ that still does not match the monomer chirality $c = 0$. Kinetics may play an important role in producing the mismatch. For stronger lateral interactions, tubules typically initiate by the formation of rings. The free energy barrier between forming a nonhelical ring and a helical one is small. Even if the initial binding yields a nonhelical ring, the single bond between a pair of monomers is weak and fluctuations can break the bond to allow the two ends to slip by each other and then bind together vertically forming a helical ring (see Fig. 9 in ref. 22). In this manner, helical tubules tend to form when $A_L > A_V$ even for the M_0^{LK} monomer. More generally, tubules with $p \neq c$ tend to form more easily for M_c^{LK} monomers at $A_L > A_V$ (see Table S2 in the ESI†).

4. Conclusions

We have studied how the interactions and molecular geometry of a simple macromolecular monomer determine the assembly into tubular structures particularly with respect to the chirality of the system. Our results have important implications in the design of macromolecular building blocks that efficiently self-assemble into tubules. In order to form tubules with a given pitch value, we introduced chirality into our model monomers such that in the ideal structure the tubule pitch would match the monomer chirality. However, we generally find that the self-assembly from free monomers into tubules with a specified pitch value does not occur. Instead a range of pitch values emerges. We show that the twist deformation stabilizes tubules with pitch inconsistent with the chirality of monomers by better aligning neighboring monomers so that they still bind well. These twisted tubules can have energy distributions that substantially overlap with that of the untwisted tubule with pitch equal to the monomer chirality. However, there is a limit to the amount of twist that yields overlapping energy distributions. Besides rotating monomers about the radial axis and aligning the lateral binding sites, twist also rotates the monomers about their vertical axis, which reduces the alignment of the vertical binding sites. Thus, to prevent twist a strong vertical interaction is required that makes protofilaments stiffer and strongly raises the cost of vertical misalignment caused by the twist. A similar trend was found recently in the packing of



filament bundles, where the ground state of a bundle of stiff filaments tend to be untwisted, while filaments with low stiffness form twisted bundles.^{29,30} However, in our case simply increasing the interaction strength between the vertical binding sites to make protofilaments stiffer does not solve the twist issue because the self-assembly of free monomers with large interaction strengths results in kinetically trapped clusters instead of tubules. In order to overcome this hurdle, we introduced a lock-and-key mechanism into our model monomers for the vertical interactions. In this manner the control of the tubule self-assembly was substantially improved and our assembly simulations were able to achieve a good match between the tubule pitch and monomer chirality.

Our results reveal the importance of the vertical interaction strength being larger than the lateral strength in microtubules.²³ This difference is necessary to prevent twist which would yield a wide range of structures that would not be biologically functional. The simulations also provide new insight into the *initial* assembly dynamics of microtubules; most experimental work has studied the growth of microtubule ends but not the initial nucleation of microtubules. Because the vertical interaction is stronger, the tubulin dimers first form protofilaments *via* vertical binding and the protofilaments subsequently bind laterally into curved sheets that close to form tubules. This route generally enhances the structural control of assembled tubules and points to the importance of precise control of the sheet-closing event. Finally, we emphasize that controlling the assembly of tubules to form a specific number of protofilaments and helicity is highly nontrivial and requires many features in the monomer. It is thus not surprising that in cells additional constraints are imposed by other molecules (*e.g.*, γ -tubulin) to achieve the degree of control observed for microtubules.

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References

- 1 S. Mann, *Nat. Mater.*, 2009, **8**, 781–792.
- 2 N. Kotov, *Science*, 2010, **330**, 188–189.
- 3 B. Alberts, A. Johnson, J. Lewis, M. Raff, K. Roberts and P. Walter, *Molecular Biology of the Cell*, Garland Science, New York, 2002.
- 4 G. D. Bachand, S. B. Rivera, A. K. Boal, J. Gaudioso, J. Liu and B. C. Bunker, *Nano Lett.*, 2004, **4**, 817–821.
- 5 R. K. Doot, H. Hess and V. Vogel, *Soft Matter*, 2007, **3**, 349–356.
- 6 H. Hess, *Annu. Rev. Biomed. Eng.*, 2011, **13**, 429–450.
- 7 Y. Tamura, R. Kawamura, K. Shikinaka, A. Kakugo, Y. Osada, J. P. Gong and H. Mayama, *Soft Matter*, 2011, **7**, 5654–5659.
- 8 A. Desai and T. Mitchison, *Annu. Rev. Cell Dev. Biol.*, 1997, **13**, 83–117.
- 9 D. Chrétien and R. H. Wade, *Biol. Cell.*, 1991, **71**, 161–174.
- 10 D. Chrétien and S. D. Fuller, *J. Mol. Biol.*, 2000, **298**, 663–676.
- 11 C. Tarabout, S. Rouxa, F. Gobeaux, N. Fayc, E. Pougeta, C. Meriadeca, M. Ligetib, D. Thomase, M. IJsselstijnb, F. Besselievreb, D.-A. Buissonb, J.-M. Verbavatzc, M. Petitjeanc, C. Valéryf, L. Perrinc, B. Rousseaub, F. Artznera, M. Paternostrec and J.-C. Cintratb, *Proc. Natl. Acad. Sci. U. S. A.*, 2011, **108**, 7679–7684.
- 12 J. Wang, H. Lu, R. Kamat, S. V. Pingali, V. S. Urban, J. Cheng and Y. Lin, *J. Am. Chem. Soc.*, 2011, **133**, 12906–12909.
- 13 J. Wang, H. Xia, Y. Zhang, H. Lu, R. Kamat, A. Dobrynin, J. Cheng and Y. Lin, *J. Am. Chem. Soc.*, 2013, **135**, 11417–11420.
- 14 J. M. Schnur, *Science*, 1993, **262**, 1669–1676.
- 15 S. Vauthey, S. Santoso, H. Gong, N. Watson and S. Zhang, *Proc. Natl. Acad. Sci. U. S. A.*, 2002, **99**, 5355–5360.
- 16 T. Shimizu, *Chem. Rev.*, 2005, **105**, 1401–1443.
- 17 T. Shimizu, *Bull. Chem. Soc. Jpn.*, 2008, **81**, 1554–1566.
- 18 Y. Forterre and J. Dumais, *Science*, 2011, **333**, 1715–1716.
- 19 C. Guo, Y. Luo, R. Zhou and G. Wei, *ACS Nano*, 2012, **6**, 3907–3918.
- 20 M. Masuda and T. Shimizu, *Chem. Commun.*, 2001, 2442–2443.
- 21 M. Masuda and T. Shimizu, *Langmuir*, 2004, **20**, 5969–5977.
- 22 S. Cheng, A. Aggarwal and M. J. Stevens, *Soft Matter*, 2012, **8**, 5666–5678.
- 23 V. VanBuren, D. Odde and L. Cassimeris, *Proc. Natl. Acad. Sci. U. S. A.*, 2002, **99**, 6035–6040.
- 24 V. Hunyadi, D. Chrétien and I. M. Jánosi, *J. Mol. Biol.*, 2005, **348**, 927–938.
- 25 F. Pampaloni, G. Lattanzi, A. Jonás, T. Surrey, E. Frey and E.-L. Florin, *Proc. Natl. Acad. Sci. U. S. A.*, 2006, **103**, 10248–10253.
- 26 V. Hunyadi, D. Chrétien, H. Flyvbjerg and I. M. Jánosi, *Biol. Cell.*, 2007, **99**, 117–128.
- 27 D. Sept and F. C. MacKintosh, *Phys. Rev. Lett.*, 2010, **104**, 018101.
- 28 J. Mozziconacci, L. Sandblad, M. Wachsmuth, D. Brunner and E. Karsenti, *PLoS One*, 2008, **3**, e3821.
- 29 R. Bruss and G. M. Grason, *Proc. Natl. Acad. Sci. U. S. A.*, 2012, **109**, 10781–10786.
- 30 R. Bruss and G. M. Grason, *Soft Matter*, 2013, **9**, 8327–8345.

