

## Single inorganic–organic hybrid nanowires with ambipolar photoresponse†

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We report for the first time single nanowires (NWs) with ambipolar (positive/negative) photoresponse that changes sign depending on the illumination wavelength. The single NWs were grown by the meniscus-guided method using inorganic (ZnO nanoparticles)–organic (PEDOT:PSS) hybrid materials. The ambipolar photoresponse of the single NWs enabled us to develop an unprecedented spectrum-discriminating NW photodetector array.

Optoelectronic devices<sup>1</sup> have attracted great attention in various domains of photonics including light generation, propagation, detection, and modulation. In particular, NW optoelectronic devices based on photoconducting properties such as NW photodetectors<sup>2,3</sup> and NW optical switches<sup>4,5</sup> are essential elements for important applications such as optical communications and optical interconnects in nanophotonic circuits.<sup>6</sup> In traditional photoconductors,<sup>7</sup> positive photoresponse is generally observed by generating mobile charge carriers in the valence and/or conduction bands under light illumination.<sup>8</sup> Ambipolar (positive/negative) photoresponse that changes sign depending on the illumination wavelength<sup>9–11</sup> or external stimuli<sup>12–14</sup> has recently emerged for multifunctional optoelectronic devices such as “light-gated” transistors,<sup>15</sup> mirror photoswitches<sup>16</sup> and logic-controllable hybrid devices.<sup>17</sup> Ambipolar photoresponse is, however, largely unexplored in single NWs that are recognized as one of the essential building blocks for nanophotonics.<sup>18</sup> Furthermore it remains a challenge to directly integrate single NWs in the designed positions<sup>19</sup> despite recent developments of flow-assisted alignment,<sup>20</sup> Langmuir–Blodgett approach,<sup>21</sup> and microcontact printing.<sup>22</sup> In this study, we report the development of single NWs with ambipolar

photoresponse based on inorganic–organic hybrid materials. We also demonstrate a novel NW photodetector array of single inorganic–organic hybrid NWs, integrated at the planned sites.

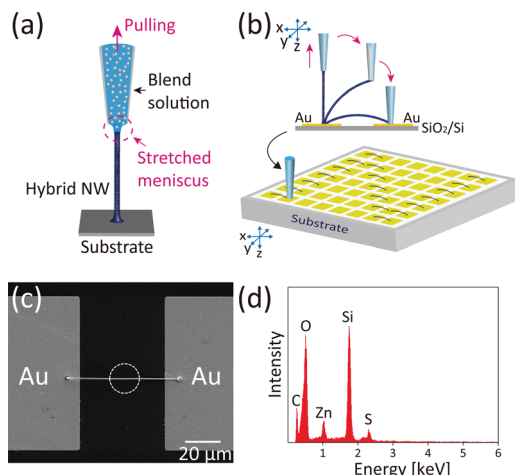
Here, blend solutions of n-type ZnO NPs<sup>23</sup> and p-type poly(3,4-ethylenedioxythiophene):poly(styrenesulfonate) (PEDOT:PSS)<sup>24</sup> prepared in water were used to grow inorganic–organic hybrid NWs. Single ZnO NP–PEDOT:PSS hybrid NWs were produced by the meniscus-guided method<sup>25,26</sup> using the blend solutions of ZnO NPs and PEDOT:PSS. The single hybrid NWs showed ambipolar photoresponse depending on the illumination wavelength. Individual growth of the single hybrid NWs at the planned sites was performed for a spectrum (UV-visible)-discriminating photodetector array. We believe that single NWs with ambipolar photoresponse will open up a novel route to design new functional nanodevices for advanced optoelectronic applications.

Fig. 1(a) schematically illustrates the growth of ZnO NP–PEDOT:PSS hybrid NWs. When a micropipette touches the substrate, a meniscus of the blend solution is created at its opening. As the micropipette (radius  $r_0 \sim 0.6 \mu\text{m}$ ) is pulled away, the meniscus is stretched, its cross-section decreases down to the nanoscale, and the solvent (water) evaporates,<sup>26</sup> yielding a hybrid NW. As the pulling continues, a freestanding, high-aspect-ratio hybrid NW is grown. The wire radius,  $r$ , can be accurately controlled by tuning the pulling speed,  $v$  (Fig. S1, ESI†). The top schematic of Fig. 1(b) shows meniscus-guided direct writing of the hybrid NWs by manipulating the micropipette using a three-axis, motion-controlled stage. To form a connected NW, we move the suspended end of the freestanding NW down to the second electrode, with the nozzle end in close proximity so that the meniscus under the nozzle extends to immerse both the NW end and the region of contact on the second electrode. Then, the micropipette is rapidly withdrawn to form the second bond.

Furthermore, we are able to precisely integrate single hybrid NWs at the planned sites by individually manipulating the micropipette and the substrate using two three-axis, motion-controlled stages (accuracy: 250 nm), as schematically

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† Electronic supplementary information (ESI) available: Experimental details, wire radius with pulling speed,  $I$ – $V$  curves of hybrid NWs, relative current densities of hybrid NWs and films, SEM images, relative current densities, and converted 8-bit color values for the hybrid NW array. See DOI: 10.1039/c3nr05910c



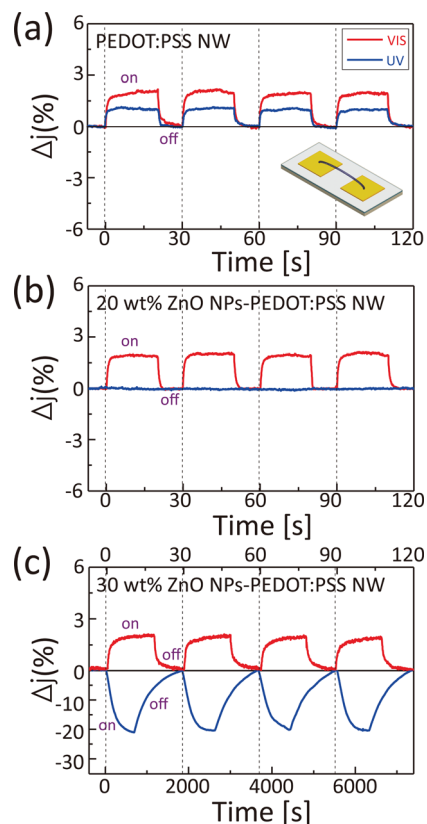
**Fig. 1** Growth of ZnO NP-PEDOT:PSS hybrid NWs. (a) Scheme of hybrid NW growth by pulling a micropipette filled with a ZnO NP-PEDOT:PSS blend solution. (b) Scheme of the precise integration of single hybrid NWs at the planned sites by meniscus-guided direct writing. (c) FE-SEM image of single hybrid NWs on Au electrodes. (d) EDS spectrum of a region (white dashed circle) of the single hybrid NW in (c).

illustrated in the bottom of Fig. 1(b). The field emission scanning electron microscopy (FE-SEM) image of Fig. 1(c) shows a single hybrid NW ( $r \sim 250$  nm and  $l$  (length)  $\sim 75$   $\mu$ m) integrated on Au electrodes. The energy-dispersive X-ray spectroscopy (EDS) spectrum in Fig. 1(d), taken from a region (white dashed circle) of the single hybrid NW, indicates the presence of C, O, Zn, and S elements (the Si peak and most of the O peaks come from the SiO<sub>2</sub>/Si substrate) from PEDOT:PSS and ZnO NPs.

The photoresponse of single hybrid NWs was dependent on ZnO NP concentration. Single hybrid NWs ( $r \sim 250$  nm,  $l \sim 75$   $\mu$ m) with various ZnO NP concentrations (0 to 50 wt%) were prepared onto pre-designed Au electrodes (inset of Fig. 2(a)). Their photoconducting properties were then measured using a two-probe method under UV ( $\lambda = 300$ –400 nm;  $0.19$  mW cm<sup>-2</sup>) or visible ( $\lambda = 400$ –700 nm;  $2.87$  mW cm<sup>-2</sup>) light illumination. All the single hybrid NWs showed ohmic behavior,<sup>27</sup> as clearly demonstrated in their  $I$ - $V$  curves (Fig. S2, ESI†). Fig. 2 shows typical variations in the relative current densities,  $\Delta j$ , recorded for three representative single NWs of (a) PEDOT:PSS, (b) 20 wt% ZnO NP-PEDOT:PSS, and (c) 30 wt% ZnO NP-PEDOT:PSS, periodically exposed to UV (or visible) light. Here  $\Delta j$  is defined as:

$$\Delta j \equiv (j_{\text{ill}} - j_0)/j_0 \quad (1)$$

where  $j_0$  and  $j_{\text{ill}}$  indicate current densities in the respective absence and presence of irradiation. For each NW, a nearly identical photoresponse is observed for multiple illumination cycles under a given light. The PEDOT:PSS NWs (0 wt% ZnO NPs) always show positive photoresponse ( $\Delta j > 0$ ) under UV and visible light illumination, as shown by the increases of  $\Delta j$  on both illumination in Fig. 2(a). The positive photoconducting behavior under visible light illumination is similarly observed



**Fig. 2** Typical variations in relative current densities,  $\Delta j$ , recorded for three representative single NWs of (a) PEDOT:PSS, (b) 20 wt% ZnO NP-PEDOT:PSS, and (c) 30 wt% ZnO NP-PEDOT:PSS, periodically exposed to UV (or visible) light. The measurements were conducted under ambient conditions at 1 V bias.

in the hybrid NWs shown in Fig. 2(b) and (c), as well as in all the hybrid NWs prepared (Fig. S3, ESI†). Under UV light illumination, however,  $\Delta j$  of the hybrid NWs decreases with ZnO NP concentration (Fig. S3, ESI†). Specifically,  $\Delta j$  drops almost to zero for 20 wt% ZnO NPs (Fig. 2(b)) and changes to negative values ( $\Delta j < 0$ ; negative photoresponse) at high concentrations ( $>20$  wt% (Fig. 2(c) and S3, ESI†). Here the change in the sign by light wavelength in Fig. 2(c) clearly indicates the ambipolar photoresponse of the hybrid NW, as also seen in the hybrid NWs at high concentrations ( $>20$  wt%) (Fig. S3, ESI†).

The photoresponse of the hybrid NWs is derived from the carriers of PEDOT:PSS (p-type semiconductor), *i.e.* holes. Under visible light illumination, electrons in the valence band of PEDOT:PSS with a band-gap energy of 1.6 eV (ref. 28) can be excited into the conduction band (see the energy diagram of Fig. S4(a), ESI†), increasing the hole carriers. This explains the positive photoresponse of the hybrid NWs under visible light illumination regardless of ZnO concentrations (Fig. 2 and S3, ESI†). On the other hand, the excitation of a ZnO NP by UV light illumination produces excited holes and electrons (Fig. S4(b), ESI†). Then, the electrons can be injected into the valence band of PEDOT:PSS (Fig. S4(b), ESI†), which decreases the carrier number of PEDOT:PSS and as a result lowers the current density  $j$  of the hybrid NWs (Fig. 2 and S3, ESI†), as suggested.<sup>10</sup>



To gain insight into the charge transfer, UV-vis absorption and FTIR studies were carried out for ZnO NP-PEDOT:PSS hybrid films that showed similar photoresponse to single hybrid NWs (compare Fig. S3 with Fig. S5, ESI†). Fig. 3(a) shows the UV-vis absorption spectra of the hybrid films as a function of ZnO NP concentration. The absorption at  $\lambda \sim 360$  nm (UV regime) is expected from the wide band-gap (3.37 eV) of ZnO.<sup>4</sup> The progressive enhancement in absorbance with ZnO NP concentration is consistent with the decrease in  $\Delta j$  under UV light illumination as shown in Fig. S5, ESI†. Here the ZnO NPs were well dispersed in single hybrid NWs, as shown in the transmission electron microscopy (TEM) images (Fig. S6, ESI†). Fig. 3(b) shows difference FTIR spectra of a hybrid film (ZnO NP concentration of 50 wt%) under UV light illumination and in the dark, acquired using the data before UV light irradiation as a baseline. The decreased absorbance at 1290, 1177, 1070, 966, and 829  $\text{cm}^{-1}$  from PEDOT<sup>29</sup> and 1004, 1047 and 1132  $\text{cm}^{-1}$  from PSS<sup>30</sup> under UV light illumination for 5 min (blue line) is, as suggested,<sup>10</sup> attributed to the reduction of PEDOT:PSS by the electron transfer *via* the excitation of ZnO NPs.<sup>10</sup> The gradual increase in the decreased absorbance in the dark (Fig. 3(b)), nearly recovering to its original state within 30 min, corresponds to the oxidation of reduced PEDOT:PSS.<sup>10</sup>

To demonstrate further the versatility of our approach, we performed individual growth of single hybrid (30 wt% ZnO NP-PEDOT:PSS) NWs with ambipolar photoresponse (Fig. 2(c)) at the planned sites, as schematically illustrated in Fig. 4(a). Fig. 4(b) shows the optical microscopy image of a “heart” pattern of the single hybrid NWs integrated on Au patterned electrodes. The single hybrid NWs were precisely positioned at the planned sites, as demonstrated by FE-SEM images shown in Fig. S7(a) and (b), ESI† for the red and blue box regions in Fig. 4(b), respectively. For each NW in Fig. 4(b), the relative current density  $\Delta j$  under visible (or UV) light illumination was measured (Fig. S8(b), ESI†) and was linearly converted to 8-bit

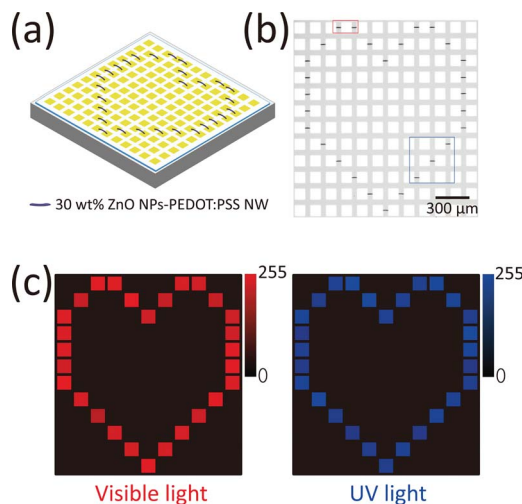


Fig. 4 Individual integration of the single hybrid (30 wt% ZnO NP-PEDOT:PSS) NW ( $r \sim 250$  nm,  $l \sim 75$   $\mu\text{m}$ ) array for a spectrum-discriminating photodetector array. (a and b) Scheme and OM image of a “heart” pattern of the single hybrid NWs on Au patterned electrodes. The red and blue box regions in (b) were observed by FE-SEM (Fig. S7(a) and (b), ESI†), respectively. (c) Ambipolar photoresponse of the single hybrid NW pattern. Each NW, as a pixel, displays 8-bit red (left) (or 8-bit blue (right)) color to which the relative current density  $\Delta j$  was linearly converted under visible (or UV) light illumination. The measurements were conducted under ambient conditions at 1 V bias.

red (or 8-bit blue) color from 0 to 255 with “255” representing the maximum (or the minimum) measured  $\Delta j$  (see eqn (1) and (2) in Fig. S8(c) and Table S1, ESI†), as demonstrated for each NW as a pixel in Fig. 4(c). We were able to discriminate visible and UV light based on the ambipolar photoresponse of the NWs, as displayed in Fig. 4(c) by the red (left) and the blue (right) colors under visible and UV light illumination, respectively. The NW-to-NW variation in  $\Delta j$  was very small,  $2.61 \pm 0.14\%$  (or  $-18.91 \pm 2.14\%$ ) (mean  $\pm$  SD) for visible (or UV) light, indicating the excellent uniformity of the integrated NWs. Single hybrid NWs with ambipolar photoresponse open an exciting opportunity for the creation of a spectrum-discriminating NW photodetector that can bridge UV photonics<sup>31</sup> and visible photonics.<sup>32</sup>

## Conclusions

In summary, we developed single ZnO NP-PEDOT:PSS hybrid NWs with ambipolar photoresponse depending on the illumination wavelength, using the meniscus-guided method. Ambipolar photoresponse was realized for the single hybrid NWs at high ZnO NP concentrations ( $>20$  wt%). By individually integrating single hybrid (30 wt% ZnO NP-PEDOT:PSS) NWs at the planned sites, we demonstrated a spectrum-discriminating NW photodetector array that can discriminate UV and visible light. The NW array showed a very small NW-to-NW photoresponse variation ( $\Delta j$ :  $2.61 \pm 0.14\%$  (or  $-18.91 \pm 2.14\%$ ) (mean  $\pm$  SD) for visible (or UV) light), indicating the excellent uniformity of integrated NWs. We believe that single NWs with ambipolar photoresponse hold a promising route for the

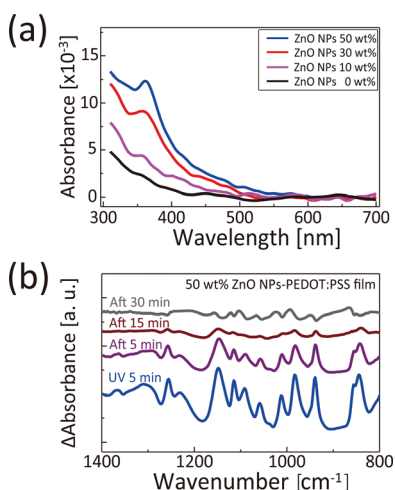


Fig. 3 UV-vis absorption and FTIR analyses. (a) UV-vis absorption spectra of ZnO NP-PEDOT:PSS hybrid films on quartz glass. (b) FTIR difference spectra of a 50 wt% ZnO NP-PEDOT:PSS hybrid film on Si wafer.



development of a variety of optoelectronic nanodevices with unusual functionality.

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## Notes and references

- 1 H. Sirringhaus, N. Tessler and R. H. Friend, *Science*, 1998, **280**, 1741; F. Bonaccorso, Z. Sun, T. Hasan and A. C. Ferrari, *Nat. Photonics*, 2010, **4**, 611; S. A. McDonald, G. Konstantatos, S. Zhang, P. W. Cyr, E. J. D. Klem, L. Levina and E. H. Sargent, *Nat. Mater.*, 2005, **4**, 138.
- 2 G. A. O'Brien, A. J. Quinn, D. A. Tanner and G. Redmond, *Adv. Mater.*, 2006, **18**, 2379.
- 3 N. Mathews, B. Varghese, C. Sun, V. Thavasi, B. P. Andreasson, C. H. Sow, S. Ramakrishnan and S. G. Mhaisalkarn, *Nanoscale*, 2010, **2**, 1984.
- 4 H. Kind, H. Yan, B. Messer, M. Law and P. Yang, *Adv. Mater.*, 2002, **14**, 158.
- 5 C.-H. Hsieh, L.-J. Chou, G.-R. Lin, Y. Bando and D. Golberg, *Nano Lett.*, 2008, **8**, 3081.
- 6 P. Yang, R. Yan and M. Fardy, *Nano Lett.*, 2010, **10**, 1529.
- 7 X. Gong, M. Tong, Y. Xia, W. Cai, J. S. Moon, Y. Cao, G. Yu, C.-L. Shieh, B. Nilsson and A. J. Heeger, *Science*, 2009, **325**, 1665; Y. Jin, J. Wang, B. Sun, J. C. Blakesley and N. C. Greenham, *Nano Lett.*, 2008, **8**, 1649; X. Xie, S.-Y. Kwok, Z. Lu, Y. Liu, Y. Cao, L. Luo, J. A. Zapien, I. Bello, C.-S. Lee, S.-T. Lee and W. Zhang, *Nanoscale*, 2012, **4**, 2914; Z. Wang, M. Safdar, C. Jiang and J. He, *Nano Lett.*, 2012, **12**, 4715.
- 8 N. V. Joshi, in *Photoconductivity: Art, Science and Technology*, ed. M. Dekker, New York, 1990, ch. 1, pp. 1–33.
- 9 W. Liu, J.-S. Lee and D. V. Talapin, *J. Am. Chem. Soc.*, 2013, **135**, 1349.
- 10 N. Sakai, G. K. Prasad, Y. Ebina, K. Takada and T. Sasaki, *Chem. Mater.*, 2006, **18**, 3596.
- 11 S. Yang, J. Gong and Y. Deng, *J. Mater. Chem.*, 2012, **22**, 24522.
- 12 J.-L. Zhu, G. Zhang, J. Wei and J.-L. Sun, *Appl. Phys. Lett.*, 2012, **101**, 123117.
- 13 H. Nakanishi, K. J. M. Bishop, B. Kowalczyk, A. Nitzan, E. A. Weiss, K. V. Tretiakov, M. M. Apodaca, R. Klajn, J. F. Stoddart and B. A. Grzybowski, *Nature*, 2009, **460**, 371.
- 14 L. Guo, X. Wang, L. Feng, X. Zheng, G. Chen, X. Yang, F. Xu, N. Tang, L. Lu, W. Ge and B. Shen, *Appl. Phys. Lett.*, 2013, **102**, 072103.
- 15 M. Kim, N. S. Safron, C. Huang, M. S. Arnold and P. Gopalan, *Nano Lett.*, 2012, **12**, 182.
- 16 S. Liu, J. Li, Q. Shen, Y. Cao, X. Guo, G. Zhang, C. Feng, J. Zhang, Z. Liu, M. L. Steigerwald, D. Xu and C. Nuckolls, *Angew. Chem., Int. Ed.*, 2009, **48**, 4759.
- 17 C. Jia, H. Li, J. Jiang, J. Wang, H. Chen, D. Cao, J. F. Stoddart and X. Guo, *Adv. Mater.*, 2013, **25**, 6752.
- 18 R. Yan, D. Gargas and P. Yang, *Nat. Photonics*, 2009, **3**, 569; P. J. Pauzauskie and P. Yang, *Mater. Today*, 2006, **9**, 36.
- 19 B. Su, S. Wang, J. Ma, Y. Wu, X. Chen, Y. Song and L. Jiang, *Adv. Mater.*, 2012, **24**, 559.
- 20 Y. Huang, X. Duan, Q. Wei and C. M. Lieber, *Science*, 2001, **291**, 630.
- 21 P. Yang, *Nature*, 2003, **425**, 243; A. R. Tao, J. Huang and P. Yang, *Acc. Chem. Res.*, 2008, **41**, 1662.
- 22 J.-H. Ahn, H.-S. Kim, K. J. Lee, S. Jeon, S. J. Kang, Y. Sun, R. G. Nuzzo and J. A. Rogers, *Science*, 2006, **314**, 1754.
- 23 A. Kudo, H. Yanagi, K. Ueda, H. Hosono, H. Kawazoe and Y. Yano, *Appl. Phys. Lett.*, 1999, **75**, 2851.
- 24 S. Sista, M.-H. Park, Z. Hong, Y. Wu, J. Hou, W. L. Kwan, G. Li and Y. Yang, *Adv. Mater.*, 2010, **22**, 380.
- 25 J. T. Kim, S. K. Seol, J. Pyo, J. S. Lee, J. H. Je and G. Margaritondo, *Adv. Mater.*, 2011, **23**, 1968.
- 26 J. T. Kim, J. Pyo, J. Rho, J.-H. Ahn, J. H. Je and G. Margaritondo, *ACS Macro Lett.*, 2012, **1**, 375.
- 27 A. J. Kronemeijer, H. B. Akkerman, T. Kudernac, B. K. van Wees, B. L. Feringa, P. W. M. Blom and B. de Boer, *Adv. Mater.*, 2008, **20**, 1467.
- 28 Y. Cao, G. Yu, C. Zhang, R. Menon and A. J. Heeger, *Synth. Met.*, 1997, **87**, 171.
- 29 C. Kvarnström, H. Neugebauer, S. Blomquist, H. J. Ahonen, J. Kankare, A. Ivaska and N. S. Sariciftci, *Synth. Met.*, 1999, **101**, 66.
- 30 Z.-S. Wang, T. Sasaki, M. Muramatsu, Y. Ebina, T. Tanaka, L. Wang and M. Watanabe, *Chem. Mater.*, 2003, **15**, 807.
- 31 D. J. Gargas, M. E. Toimil-Molares and P. Yang, *J. Am. Chem. Soc.*, 2009, **131**, 2125; Q. Zhuang, Z. Yang and J. Kang, *Appl. Phys. Lett.*, 2013, **102**, 171601; H. Wu, Y. Sun, D. Lin, R. Zhang, C. Zhang and W. Pan, *Adv. Mater.*, 2009, **21**, 227.
- 32 Z. Liu, L. Yin, H. Ning, Z. Yang, L. Tong and C.-Z. Ning, *Nano Lett.*, 2013, **13**, 4945; M. Law, D. J. Sirbully, J. C. Johnson, J. Goldberger, R. J. Saykally and P. Yang, *Science*, 2004, **305**, 1269; T. Zhai, X. Fang, M. Liao, X. Xu, L. Li, B. Liu, Y. Koide, Y. Ma, J. Yao, Y. Bando and D. Golberg, *ACS Nano*, 2010, **4**, 1596.

