### Dalton Transactions



PAPER View Article Online
View Journal | View Issue



**Cite this:** *Dalton Trans.*, 2014, **43**, 12776

Received 28th May 2014, Accepted 24th June 2014 DOI: 10.1039/c4dt01569j

www.rsc.org/dalton

# The structure and energetics of arsenic(III) oxide intercalated by ionic azides†

Piotr A. Guńka,\*a Karol Kraszewski, Yu-Sheng Chenb and Janusz Zachara

Unprecedented intercalates of arsenic(III) oxide with potassium azide and ammonium azide have been obtained and characterized by single crystal X-ray diffraction. The compounds are built of  $As_2O_3$  sheets separated by charged layers of cations and azide anions perpendicular to the sheets. The intercalates are an interesting example of hybrid materials whose structure is governed by covalent bonds in two directions and ionic bond in the third one. The obtained compounds are the first examples of  $As_2O_3$  intercalates containing linear pseudohalogen anions. Periodic DFT calculations of interlayer interaction energies were carried out with the B3LYP-D\* functional. The layers are held together mainly by ionic bonds, although the computations indicate that interactions between cations and  $As_2O_3$  sheets also play a significant role. A comparison of cation and anion interaction energies with neutral  $As_2O_3$  sheets sheds light on the crystallisation process, indicating the templating effect of potassium and ammonium cations. It consist in the formation of sandwich complexes of cations with crown-ether-resembling  $As_6O_{12}$  rings. Raman spectra of both compounds are recorded and computed *ab initio* and all vibrational bands are assigned.

#### Introduction

Intercalation compounds result from inclusion, without covalent bonding, of molecules or ions in a solid matrix of another compound which has a layered structure. Intercalates of various inorganic hosts exhibit a wide range of interesting properties and find numerous everyday applications. Lithium intercalates of transition-metal-based materials such as LiMnO2, LiCoO2 and FePO4 are utilised as electrodes in lithium-ion batteries, 2,3 whereas iron arsenide and iron selenide based intercalates with metals or with metals and organic solvents can be superconducting at temperatures as high as ~50 K.4-8 The alkali metal intercalates of graphite are used as reducing agents in chemical synthesis.9 There are a number of arsenic(III) oxide intercalates described in the literature whose crystal structures have been determined. These include hexagonal  $KX \cdot 2As_2O_3$  where X = Cl, Br, I;  $NH_4Y \cdot 2As_2O_3$  where Y = Br, hydrated 2NH<sub>4</sub>Cl·4As<sub>2</sub>O<sub>3</sub>·H<sub>2</sub>O<sup>11</sup> and orthorhombic NaBr·2As<sub>2</sub>O<sub>3</sub>. <sup>12</sup> To the best of our knowledge, there are no publications reporting intercalates with other types of ions, particularly non-spherical ones. We have, therefore, attempted to

## Experimental and computational details

All chemicals used, except for NH<sub>4</sub>AsO<sub>2</sub> which was synthesized as described previously,<sup>13</sup> were obtained from commercial sources and were used as received without further purification. Reactions were carried out in open glass vessels.

#### $NH_4N_3 \cdot 2As_2O_3$ (1)

 $\mathrm{NH_4AsO_2}$  crystals were stored at 5 °C after the synthesis in a closed vessel in a saturated aqueous solution of ammonium arsenite. 75 mg of the crystals were taken together with 0.5 mL of the solution and were mixed with a solution of 98 mg of  $\mathrm{NaN_3}$  in 3 mL of water. The mixture was stirred until all of the ammonium arsenite was dissolved and the resulting solution was left for crystallisation via slow water evaporation. When a third of the initial solution volume was left in the vessel,

synthesise arsenic(III) oxide intercalates with salts containing ions with different geometry and herein we report the synthesis and crystal structures of the first As<sub>2</sub>O<sub>3</sub> intercalates containing linear pseudohalogen anions. In order to understand the contribution of various components to the interlayer interaction energies, we have performed periodic DFT computations with the B3LYP-D\* functional. Moreover, we show experimental Raman spectra of the intercalates with all the bands assigned by comparison with theoretical vibrational frequencies.

<sup>&</sup>lt;sup>a</sup>Warsaw University of Technology, Faculty of Chemistry, Noakowskiego 3, 00-664 Warszawa, Poland. E-mail: piogun@ch.pw.edu.pl

<sup>&</sup>lt;sup>b</sup>University of Chicago, ChemMatCARS beamline, Advanced Photon Source, Argonne, Illinois 60439. USA

 $<sup>\</sup>dagger$  Electronic supplementary information (ESI) available: Derived formulae for interlayer interaction energies, powder diffraction patterns and Raman spectra of compounds 1 and 2, detailed results of geometry optimisations and frequency calculations. See DOI: 10.1039/c4dt01569j

Open Access Article. Published on 25 June 2014. Downloaded on 12/5/2025 9:12:55 AM.

crystallisation was stopped yielding single crystals of

crystallisation was stopped yielding single crystals of compound 1 exclusively with a yield of 60%.

#### $KN_3 \cdot 2As_2O_3(2)$

**Dalton Transactions** 

0.4 g of  $As_2O_3$  powder (2 mmol) was added to 3 mL of 2 M KOH solution (6 mmol) and stirred until all of the  $As_2O_3$  was dissolved. Afterwards, 0.13 g of  $NaN_3$  (2 mmol) was dissolved in the solution. Then, concentrated HNO $_3$  solution was added dropwise until a solid phase was precipitated (pH  $\sim$  9–10 was achieved at this point). The resulting powder was filtered and washed with a mixture of methanol and concentrated HNO $_3$  (9:1, v/v) twice. The powder was dried on filter paper. Such a procedure leads to a pure intercalate 2 with a yield of 65%. In order to grow a single crystal suitable for the X-ray diffraction experiment, the solution was diluted five times before the addition of HNO $_3$ . Nonetheless, all attempts led to tiny, plate-shaped single crystals whose diffraction pattern was only measurable using the synchrotron X-ray source.

Suitable single crystals were selected under a polarizing microscope, mounted in inert oil and transferred to the cold gas stream of the diffractometer. Diffraction data for compound 1 were measured at RT with graphite-monochromated Mo-Kα radiation on the Oxford Diffraction κ-CCD Gemini A Ultra diffractometer. Absorption effects were corrected analytically. 14 Cell refinement and data collection as well as data reduction and analysis were performed with the Oxford Diffraction software CrysAlis PRO. 15 Diffraction data for compound 2 were collected at ChemMatCARS beamline, 15-ID-B, at the Advanced Photon Source (APS), USA. A Bruker APEXII CCD detector was used to record the diffracted intensities at  $\lambda$  = 0.38745 Å (32 keV) and at the following temperatures: 45(2), 35(2), 25(2), 15(2) and 30(2) K. Data reduction and analysis including multi-scan absorption and oblique corrections were carried out using the APEX-II suite and SADABS. 16 The structures were solved by direct methods and subsequent Fourierdifference synthesis and refined by full-matrix least-squares against F2 with SHELX-201317 within the Olex2 suite which was also used for data analysis. 18 Ammonium cations in compound 1 lie on a special position exhibiting 6/mmm site symmetry and hydrogen atoms are disordered. Figures of crystal structures were created using Diamond<sup>19</sup> and rendered with POV-Ray.<sup>20</sup> All of the discussions on the crystal structure of compound 2 relate to the 45 K measurement, if not stated otherwise.‡

‡ Crystal data for compound 1:  $2\text{As}_2\text{O}_3$ ·NH<sub>4</sub>N<sub>3</sub>,  $M = 455.76 \text{ g mol}^{-1}$ , a = 5.2354(3) Å, b = 5.2354(3) Å, c = 9.6787(5) Å,  $\alpha = 90^{\circ}$ ,  $\beta = 90^{\circ}$ ,  $\gamma = 120^{\circ}$ , V = 229.75(3) Å<sup>3</sup>, T = 293(2) K, space group P6/mmm, Z = 1,  $\mu(\text{MoK}\alpha) = 14.432 \text{ mm}^{-1}$ , 7048 reflections measured, 171 independent reflections ( $R_{\text{int}} = 0.0493$ ). The final  $R_1$  and  $wR(F^2)$  values ( $I > 2\sigma(I)$ ) were 0.0155 and 0.0379, respectively. The final  $R_1$  and  $wR(F^2)$  values (all data) were 0.0192 and 0.0388, respectively. The goodness of fit on  $F^2$  was 1.220.

Crystal data for compound 2:  $2\text{As}_2\text{O}_3$ ·KN<sub>3</sub>,  $M = 476.81 \text{ g mol}^{-1}$ , a = 5.2425(10) Å, b = 5.2425(10) Å f, c = 9.4865(17) Å,  $a = 90^\circ$ ,  $\beta = 90^\circ$ ,  $\gamma = 120^\circ$ ,  $V = 225.79(10) \text{ Å}^3$ , T = 45(2) K, space group P6/mmm, Z = 1,  $\mu$ (synchrotron) = 2.872 mm<sup>-1</sup>, 775 reflections measured, 154 independent reflections ( $R_{\text{int}} = 0.0236$ ). The final  $R_1$  and  $wR(F^2)$  values ( $I > 2\sigma(I)$ ) were 0.0220 and 0.0538, respectively. The final  $R_1$  and  $wR(F^2)$  values (all data) were 0.0225 and 0.0539, respectively. The goodness of fit on  $F^2$  was 1.356.

Raman spectra were recorded using a Nicolet Almega Dispersive Raman spectrometer. Spectra of powders of 1 and 2 were obtained using a 780 nm excitation line and a 1200 lines mm<sup>-1</sup> resolution grating. The exposition time was 30 s.

Periodic quantum mechanical computations were carried out using the CRYSTAL09 programme suite. 21-23 Calculations were performed within the DFT<sup>24,25</sup> framework using the hybrid B3LYP functional26 with an all-electron TZVP basis set optimised for solid state computations (pob-TZVP).27 Dispersion interactions were accounted for by an empirical correction scheme proposed by Grimme<sup>28</sup> with a modified parameterisation developed by Civalleri et al. (B3LYP-D\* model).29 The five tolerances setting the accuracy of the Coulomb and exchange series were arbitrarily set to 7, 7, 7, and 25 and SCF convergence on the total energy was set to 10<sup>-7</sup> and 10<sup>-10</sup> hartree for geometry optimisation and for frequency calculations, respectively. The shrinking factor in the reciprocal net was set to 8, an extra-large grid was used for charge integration (XLGRID keyword) and a modified Broyden scheme,<sup>30</sup> following the method proposed by Johnson,<sup>31</sup> was used to speed up the SCF convergence. Vibrations were visualized and analysed using J-ICE.32

#### Results and discussion

#### **Crystal structures**

The first report on the crystal structure of an arsenic(III) oxide intercalate was published in the 1950s. It was a hexagonal hydrated intercalate of ammonium chloride.11 Afterwards, in the 1980s Pertlik obtained a series of intercalates with ammonium and potassium halides as well as sodium bromide by means of hydrothermal crystallisations. All of the compounds exhibited hexagonal symmetry except for the intercalate with NaBr which crystallised in the orthorhombic crystal system. 10,12 We have obtained new intercalates from arsenic (III) oxide dissolved in ammonia or potassium hydroxide aqueous solutions containing sodium azide. Crystallisations were effected by RT evaporation of water in the case of ammonia solution or by addition of a mineral acid to the KOH solution. In both cases only one crystalline phase was obtained which was confirmed by the comparison of samples' experimental powder diffractograms with ones generated from the determined crystal structures (see Fig. S1 in the ESI†). Similarly to the so-far-known intercalates with potassium and ammonium halides, intercalates with NH<sub>4</sub>N<sub>3</sub> (1) and KN<sub>3</sub> (2) crystallise in the P6/mmm space group and their crystal structures are isotypic (see Fig. 1). The As-O bond lengths equal 1.7902(13) Å in compound 1 and 1.793(2) Å in 2 which agrees well with the values reported by Pertlik. For example, As-O distances in NH<sub>4</sub>Br and KCl intercalates of As<sub>2</sub>O<sub>3</sub> amount to 1.801(1) and 1.793(2) Å, respectively. The AsO<sub>3</sub> ψ-tetrahedra share all oxygen ligands, forming neutral As<sub>2</sub>O<sub>3</sub> sheets, oriented parallel to the (001) lattice planes. The topology of the As<sub>2</sub>O<sub>3</sub> layers is the same as that in other hexagonal intercalates with potassium and ammonium halides. They are planar and all of the **Paper** 

Fig. 1 Crystal structures of intercalates 1 and 2. View along the [001] direction in a ball and stick model and view along the [120] direction with ADPs visualised are presented at the top and at the bottom, respectively. Thermal ellipsoids are drawn at the 50% probability level.

arsenic atomic cores are situated on one side of the plane formed by oxygen ligands. The oxygen side of the sheets is adjacent to a layer of cations and, the other side, where pendent arsenic lone electron pairs (LEPs) are situated, interacts with azide anions perpendicular to the sheets. This results in a periodic pattern of neutral As<sub>2</sub>O<sub>3</sub> sheets and alternating ionic layers (see Fig. 1). The As...N separations are 3.178(2) Å for 1 and 3.189(5) for 2 which is a little less than the sum of arsenic and nitrogen van der Waals radii (3.40 Å), indicating the presence of weak interactions between arsenic atoms and azide anions. Similarly to intercalates with potassium and ammonium halides, the threefold coordination of arsenic atoms by oxygen ligands is complemented by three N<sub>3</sub><sup>-</sup> anions, forming a distorted octahedron. A survey of CSD reveals that As...N<sub>3</sub> contacts shorter than the sum of vdW radii are rather rare. Only one compound with a separation of 2.973(3) Å was found (refcode HUGLIE). Nonetheless, there are reports on compounds containing As(III), Sb(III) and Bi(III) coordination centres and azide ligands.33-36 The azide anions are symmetrical with N-N bond lengths of 1.173(7) and 1.161(16) Å for compounds 1 and 2, respectively. Such bond lengths are typical of ionic inorganic azides.<sup>37</sup>

Due to the separation of ions having opposite charge by neutral  $As_2O_3$  sheets, the interionic  $K\cdots N_{middle}$  and  $N\cdots N_{middle}$  distances of 4.7433(9) and 4.8393(3) Å are much longer than in the  $KN_3$  and  $NH_4N_3$  ionic crystals (3.5372(3) and 3.740(2),

respectively).<sup>37,38</sup> There are quite large voids in the crystal structures, located in ionic layers. The packing indices for compounds **1** and **2** equal 80.9% and 88.1%, respectively.

2

When carrying out diffraction experiments at the APS synchrotron, we noted that the crystals of the intercalate 2 broke upon flash-cooling to 15 K which was manifested by the presence of split reflections in their diffraction patterns. Nevertheless, the intercalate crystals remained intact when cryo-cooled to 45 K. Suspecting a polymorphic transition, we decided to undertake a multi-temperature study of the intercalate's crystal structure. We recorded the diffraction patterns of one crystal at the following temperatures in the given order: 45, 35, 25, 15 and 30 K. Reflections recorded at 45 K were not split and cooling the crystal by 10 °C did not cause any division of reflections. We did not observe any polymorphic transition either. However, the unit cell was expanding as a function of irradiation time (equivalent to the X-ray dose) and the increase of displacement parameters for terminal nitrogen atoms of azide anions was much more pronounced than for all the other atoms (Fig. S2 and Table S1 in the ESI†). Expansion of the unit cell and an increase of  $B_{iso}$  value with accumulating X-ray dose are commonly observed symptoms of radiation damage in protein crystallography.39 This suggests that the studied crystal was subject to radiation damage which proceeded mainly via decomposition of azide anions, probably accompanied by nitrogen evolution.

**Dalton Transactions** 

#### **Interlayer interaction energies**

DFT computations with periodic boundary conditions were carried out in the CRYSTAL09 program suite in order to understand the energetic factors responsible for the formation of a layered structure of intercalates. Both the energies of crystal lattices and the energies of electrically neutral layers extracted from 3D periodic structures were computed. The B3LYP functional was chosen, as it is known to describe oxidic and semiconducting solids accurately40 and to predict vibrational spectra of inorganic systems well.41 We have also used the Grimme correction for the dispersion forces (DFT-D2), 28 as we expected it to be significant in the case of the studied materials.<sup>29</sup> The space group symmetry of compound 1 was arbitrarily lowered to P31m for computations to circumvent the disorder of ammonium hydrogen atoms. The geometry of layers extracted for interaction energy calculations was kept fixed at the optimised geometry of periodic structures. Two strata of ghost atoms were added to each side of the extracted layers when BSSE was estimated according to the BB-CP method.42

There are several possible options of extracting electrically neutral layers from the studied intercalates (Fig. 2). The choices may be divided into three groups: layers terminated with an ion on one side only, denoted a and b, layers terminated with ions on both sides, denoted c and d, and purely arsenic(III) oxide layers. The a and b choices of layers as well as the c and d choices are related by the inversion centre. They are unique in the axial P31m space group utilised for compound 1 computations and equivalent by symmetry in the

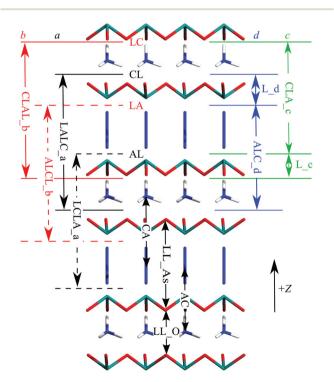


Fig. 2 Possible layer extraction choices and interlayer interaction energies in crystal structures of intercalates 1 and 2.

centrosymmetric P6/mmm space group of compound 2 which reduces the number of possible extractions by two. There are four pairs of interlayer interactions present in the investigated structures - the interactions between cations (C) and As<sub>2</sub>O<sub>3</sub> neutral layers (L) with the interaction energies denoted  $E_{CL}$ and  $E_{\rm LC}$  (the order of letters is consistent with the -Z direction); the interactions of anions (A) with  $As_2O_3$  layers:  $E_{AL}$  and  $E_{LA}$ ; ionic interactions  $E_{CA}$  and  $E_{AC}$ ; and the interactions between neutral As<sub>2</sub>O<sub>3</sub> sheets, which we will refer to as  $E_{\text{LL, AS}}$ when layers are chosen such that arsenic atoms from adjacent sheets point towards one another and  $E_{\text{IL},O}$  otherwise. To compute the interaction energies, we note that unit cell energy can be decomposed into the following terms when layers are extracted according to choices c and d:

$$\varepsilon_0 = \varepsilon_{\text{CLA\_c}} + \varepsilon_{\text{L\_c}} + E_{\text{AC}} + E_{\text{LL\_O}} + E_{\text{LL\_AS}} + E_{\text{AL}} + E_{\text{LC}}$$
 (1a)

$$\varepsilon_0 = \varepsilon_{\text{ALC\_d}} + \varepsilon_{\text{L\_d}} + E_{\text{CA}} + E_{\text{LL\_O}} + E_{\text{LL\_AS}} + E_{\text{LA}} + E_{\text{CL}}$$
 (1b)

where  $\varepsilon_0$  denotes unit cell energy,  $\varepsilon_{\rm I~i}$  denotes energy of an extracted layer of sequence I corresponding to choice i and  $E_{\text{INT}}$  is the interaction energy INT (see Fig. 2 for the clarification of the notation). We find analogous decomposition equations for the choices a and b:

$$\varepsilon_0 = \varepsilon_{\text{LALC}\_a} + E_{\text{CL}} + E_{\text{CA}} + E_{\text{LL}\_O} \tag{2a}$$

$$\varepsilon_0 = \varepsilon_{\text{LCLA}\_a} + E_{\text{AL}} + E_{\text{AC}} + E_{\text{LL}\_\text{As}} \tag{2b}$$

$$\varepsilon_0 = \varepsilon_{\text{CLAL\_}b} + E_{\text{LC}} + E_{\text{AC}} + E_{\text{LL\_O}} \tag{2c}$$

$$\varepsilon_0 = \varepsilon_{\text{ALCL}\_b} + E_{\text{LA}} + E_{\text{CA}} + E_{\text{LL}\_As} \tag{2d}$$

In order to estimate the  $E_{\rm LL\_O}$  and  $E_{\rm LL\_As}$  we extract only the As<sub>2</sub>O<sub>3</sub> sheets from intercalate structures and we notice that:

$$\varepsilon_{\text{LL\_O}} = \varepsilon_{\text{L\_c}} + \varepsilon_{\text{L\_d}} + E_{\text{LL\_O}} \tag{3a}$$

$$\varepsilon_{\text{LL\_As}} = \varepsilon_{\text{L\_c}} + \varepsilon_{\text{L\_d}} + E_{\text{LL\_As}}$$
 (3b)

Note that the following pairs of equations are equivalent by symmetry for compound 2: (1a) and (1b), (2a) and (2c) as well as (2b) and (2d). The solution of the system of linear equations is straightforward and is given in Table S2 in the ESI.† Obtained interaction energy values are presented in Table 1. The optimised structures of compounds 1 and 2 as well as  $\varepsilon_{\rm I}$  i values are given in Tables S3 and S4.† The BSSE error was estimated using the BB-CP method, which in practice consist in adding ghost atoms to the extracted layers where additional basis functions are centered. 42 As can be seen from Table 1, the BSSE error is huge for anion---layer interactions and substantial for cation...layer contacts. This observation agrees well with the conclusion of Brandenburg and co-workers presented in a recent publication where a geometric method of BSSE correction has been extended to periodic DFT calculations. 43 It was shown there that BSSE can be enormous for the pob-TZVP basis set.

As expected the  $E_{CA}$  and  $E_{AC}$  electrostatic energies for both compounds are much larger than all the other interlayer interaction energies, indicating that it is the ionic bond that is mainly responsible for holding the layers together in 1 and 2

**Table 1** Computed interlayer interaction energies for compounds 1 and 2. E stands for interaction energy,  $E^C$  is the BSSE-corrected value and  $E_{\rm ion}$  is the electrostatic energy of the ionic bond estimated from the Madelung approximation with the EUGEN code. All values are given in kJ mol<sup>-1</sup> per unit cell

	Compound	Interlayer interaction	E	$E^{\rm C}$	$E_{ m ion}$
1	NH <sub>4</sub> N <sub>3</sub> ·2As <sub>2</sub> O <sub>3</sub>	CA	-237.5	-233.4	$-281.1^{a}$
	P31m	AC	-237.4	-234.6	$-272.3^{b}$
		LL_O	-24.9	-24.8	
		LL_As	-14.0	-15.5	
		CL	-55.0	-41.7	
		LC	-37.3	-26.1	
		AL	-30.6	-10.7	
		LA	-30.3	-11.4	
2	$KN_3 \cdot 2As_2O_3$	CA	-240.1	-234.9	$-289.3^{a}$
	P6/mmm	LL_O	-24.9	-25.1	$-273.0^{b}$
		LL_As	-14.1	-18.3	
		CL	-68.6	-51.7	
		AL	-32.4	-12.1	

 $^a$  Both ions were treated as monopoles.  $^b$  Cations were treated as monopoles and azide anions as two -0.5 partial point charges located on terminal nitrogen atoms.

intercalate crystals. The energy is dependent only on interacting ion charges and their spatial distribution, being independent of the ions' nature. These energies are essentially the same because they can be treated as the Madelung energies of the crystals. 44-47 The Madelung constants for 1 and 2 are equal as both compounds crystallise in the same invariant crystal structure type. The Madelung energy, on the other hand, depends also on the interionic distances and, hence, the observed differences. 48 We have computed the electrostatic energy of the ionic NH<sub>4</sub>N<sub>3</sub> and KN<sub>3</sub> lattices obtained from the optimised intercalate 1 and 2 structures by removing As<sub>2</sub>O<sub>3</sub> sheets. We have used a method for calculating Madelung constants implemented in the EUGEN code. 49 The electrostatic energy values are given in Table 1. The absolute values obtained from periodic DFT calculations are smaller than the energy evaluated with the EUGEN code, as the DFT values take into account screening by the As<sub>2</sub>O<sub>3</sub> sheets. These values are also significantly lower than the lattice energies of ionic NH<sub>4</sub>N<sub>3</sub> and KN<sub>3</sub> crystals which amount to -670.3 and -673.2 kJ mol<sup>-1</sup>, respectively (all ions were treated as point charges and crystal structures were taken from ref. 37 and 38). The crystallisation of intercalates causes spatial separation of cations and anions by neutral As2O3 sheets, leading to a substantial decrease of ionic bond strength. Nonetheless, the studied compounds are an interesting example of hybrid materials whose structures are governed by covalent As-O bonds in two dimensions (ab planes) and by ionic bond in the third direction (Z axis).

The  $E_{\rm CL}$  and  $E_{\rm LC}$  energies for compound 1 differ significantly from each other, which stems from the fact that different interlayer hydrogen bond motifs are cut in the two cases. In the case of the CL boundary, there are three hydrogen atoms pendant from cationic layers, whereas there is only one in the case of the LC boundary. The hydrogen atom is in addition less favourably located for hydrogen bond formation

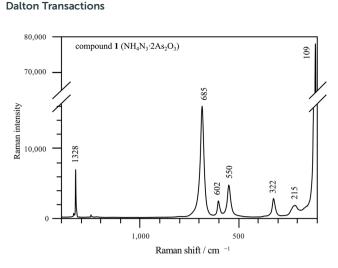
than the other three. This results in an interaction energy smaller by 15.6 kJ  $\text{mol}^{-1}$  for boundary LC. An analogous  $|E_{\text{AL}} - E_{\text{LA}}|$  energy difference amounts to as little as 0.7 kJ  $\text{mol}^{-1}$ . Surprisingly, the C···L interaction energies are higher for potassium than for ammonium cations, while A···L interaction energies are very similar in 1 and 2. It is important to realise that the C···L interlayer interaction energies are much higher than the A···L interaction energies, which points to stronger cation interactions with As<sub>2</sub>O<sub>3</sub> sheets.

Closer inspection of the 12-fold coordination sphere of potassium and ammonium cations with K···O and N···O distances of 3.0790(19) and 3.1349(14) Å, respectively, reveals its resemblance to cations' coordination by 18-crown-6 ethers, where the K···O and N···O average separations are 2.85(8) and 2.97(7) Å, respectively.50 Cation...oxygen distances are lower for the complexes with crown ethers as cations enter the cavity of the chelating ligands while they are sandwiched between two crownether-like As<sub>6</sub>O<sub>12</sub> rings in the intercalates. This and the fact that interaction energies with cations are higher than that with anions may indicate a templating effect of cations during intercalates crystallisation. It might be assumed that sandwich complexes of cations with six-membered As<sub>6</sub>O<sub>12</sub> rings are formed at the first stage, which in turn condense to yield As<sub>2</sub>O<sub>3</sub> layers. All of the so-far-studied arsenic(III) oxide intercalates containing potassium and ammonium cations crystallise with As<sub>2</sub>O<sub>3</sub> sheets displaying the same geometry. <sup>10,11</sup> The only intercalate structure with different As<sub>2</sub>O<sub>3</sub> sheet geometry is the one with NaBr, confirming the decisive role of cations in the shaping of arsenic(III) oxide sheets. 12 However, our attempts to obtain an intercalate with NaN3 in an analogous procedure to the one described above proved unsuccessful.

#### Raman spectra

We have recorded Raman spectra of both compounds (Fig. 3) and performed quantum mechanical calculations to assign the observed bands to normal modes for compound 2 (see Table 2).23,41 The spectrum was evaluated computationally at the  $\Gamma$  point using harmonic approximation. The polarised band located at 1329 cm<sup>-1</sup> in the experimental spectrum corresponds to symmetric stretching of azide anions, whereas the remaining bands are due to As<sub>2</sub>O<sub>3</sub> layer vibrations. The bands located at 686, 604, 554, 321 and 217 cm<sup>-1</sup> result from deformation vibrations of AsO<sub>3</sub> ψ-tetrahedra and the peak at 113 cm<sup>-1</sup> corresponds to the vibrations of the whole As<sub>2</sub>O<sub>3</sub> layers with respect to a rigid network of KN3. Fig. 4 shows eigenvectors associated with all normal modes of As2O3 layer vibrations and corresponding eigenvalues (harmonic frequencies). The agreement of the observed vibration frequencies with the theoretically predicted ones is excellent with the highest difference of 32 cm<sup>-1</sup> (for the 185 cm<sup>-1</sup> band) and the r.m.s. deviation of 14 cm<sup>-1</sup>.

The Raman spectrum of compound 1 contains a peak at  $3190~\text{cm}^{-1}$  which we attribute to N–H vibrations. The bands corresponding to  $As_2O_3$  layer vibrations as well as to the symmetric stretching of azide anions are present in the spectrum of compound 1 and they are located at the same frequencies as



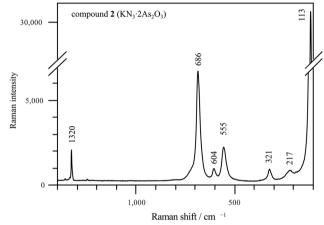


Fig. 3 Raman spectra of compounds 1 and 2. Spectra in the whole registered wavenumber range are depicted in Fig. S3 in the ESI. $\dagger$ 

Table 2 Assignment of compound 1 and 2 Raman bands (cm<sup>-1</sup>)

	$1^a$		2		
		$\overline{\text{NH}_4\text{N}_3 \cdot 2\text{As}_2\text{O}_3} \qquad \overline{\text{KN}_3 \cdot 2\text{As}_2\text{O}_3}$			
	Symmetry	Exp	B3LYP-D*	Exp	Assignment
1	$A_{1g}$	109 p	97	113 р	$\delta_{ m sAsO3}$
2	$E_{2g}$	215 p	185	217 p	$\delta_{ m sAsO3}$
3	$E_{1g}$	322 p	318	321 p	$\delta_{\mathrm{sAsO3}}$
4	$E_{2g}$	550 p	555	555 p	$\delta_{ m sAsO3}$
5	$E_{1g}$	602 p	607	604 p	$\delta_{ m sAsO3}$
6	$A_{1g}$	685 p	682	686 p	$\delta_{ m sAsO3}$
7	$A_{1g}$	1328 p	1323	1329 p	$\nu_{ m sN3}$
8		3190 p			N-H stretching

<sup>&</sup>lt;sup>a</sup> p stands for polarized.

in compound 2 within a few inverse centimetres range (see Table 2).

This remarkable agreement is understandable given that the exchange of potassium for ammonium cation exercises a rather weak influence on  $As_2O_3$  sheets. Consequently, we expect the frequencies of the  $As_2O_3$  layer deformations to remain largely unchanged as different kinds of anions and

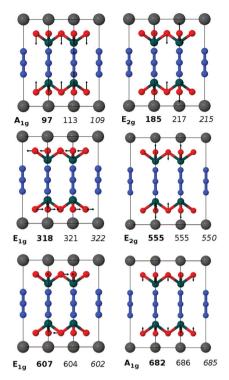


Fig. 4 Atomic displacements for normal modes of  $As_2O_3$  sheets as calculated for intercalate 2. The calculated harmonic frequencies as well as symmetry are given in bold, and experimental frequencies for compounds 1 and 2 are given in normal font and italics, respectively.

cations are introduced into the crystal structure as long as the symmetry and topology of layers are retained.

The assignment for compound 1 was not attempted by comparison with the computationally predicted spectrum as calculations were carried out in the *P*31*m* space group. Lowering of the symmetry, especially removal of the inversion centre, has a significant impact on the spectrum and on the symmetry of vibrations, making comparison with experimental spectra very difficult. See Fig. S3 and Tables S5 and S6 in the ESI† for more results concerning the frequency calculations of intercalates 1 and 2.

#### Conclusions

The synthesis and structural characterisation of the first  $As_2O_3$  intercalates with a non-spherical anion are presented. The obtained compounds containing linear pseudohalogen azide anion are isotypic with the intercalates containing spherical halides. Spatial separation of ions by neutral  $As_2O_3$  layers has been achieved and  $As\cdots N_3^-$  contacts have been observed. Periodic DFT computations of interlayer interaction energies as well as the structural similarity of the coordination of  $K^+$  and  $NH_4^+$  cations by  $As_2O_3$  sheets with their coordination by crown ethers indicates the templating effect of cations on arsenic(III) oxyanions and sheds light on the crystallisation process. It is the cations' interaction with  $As_6O_{12}$  cyclic units that is the decisive factor for the planar hexagonal  $As_2O_3$  sheets formation. It

has been confirmed by the computations that the ionic bond is mainly responsible for holding layers together in crystals. The studied compounds are, therefore, an example of hybrid materials whose structure is governed by covalent bonds in two directions and by ionic bond in the third one. Last but not least, the Raman spectrum of the intercalate 1 has been successfully predicted *ab initio* and all of the bands in both studied compounds have been assigned. The agreement between calculated and observed frequencies is very good.

#### Acknowledgements

**Paper** 

This work was supported by the "Iuventus Plus" program of the Polish Ministry of Science and Higher Education (0242/ IP3/2013/72). Calculations have been carried out using resources provided by the Wroclaw Centre for Networking and Supercomputing (http://wcss.pl), grant no. 260. ChemMatCARS Sector 15 is principally supported by the National Science Foundation/Department of Energy under grant number NSF/ CHE-0822838. The use of the Advanced Photon Source, an Office of Science User Facility operated for the U.S. Department of Energy (DOE) Office of Science by Argonne National Laboratory, was supported by the U.S. DOE under contract no. DE-AC02-06CH11357. Raman spectra acquisition Z. Żukowska, provision of the modified version of the EUGEN code by S. Tan and E. I. Pas (Izgorodina) and critical comments from M. Lesiuk are gratefully acknowledged.

#### Notes and references

- 1 A. D. McNaught and A. Wilkinson and International Union of Pure and Applied Chemistry, *Compendium of chemical terminology: IUPAC recommendations*, Blackwell Science, Oxford, England, Malden, MA, USA, 1997.
- 2 A. K. Padhi, K. S. Nanjundaswamy and J. B. Goodenough, J. Electrochem. Soc., 1997, 144, 1188–1194.
- 3 M. Winter, J. O. Besenhard, M. E. Spahr and P. Novák, *Adv. Mater.*, 1998, **10**, 725–763.
- 4 M. Rotter, M. Tegel and D. Johrendt, *Phys. Rev. Lett.*, 2008, 101, 107006.
- 5 Z. A. Ren, J. Yang, W. Lu, W. Yi, G. C. Che, X. L. Dong, L. L. Sun and Z. X. Zhao, *Mater. Res. Innov.*, 2008, 12, 105–106.
- 6 T. P. Ying, X. L. Chen, G. Wang, S. F. Jin, T. T. Zhou, X. F. Lai, H. Zhang and W. Y. Wang, Sci. Rep., 2012, 2.
- 7 A. Krzton-Maziopa, E. V. Pomjakushina, V. Y. Pomjakushin, F. von Rohr, A. Schilling and K. Conder, *J. Phys.: Condens. Matter*, 2012, 24, 382202.
- 8 A. Krzton-Maziopa, Z. Shermadini, E. Pomjakushina, V. Pomjakushin, M. Bendele, A. Amato, R. Khasanov, H. Luetkens and K. Conder, *J. Phys.: Condens. Matter*, 2011, 23, 052203.
- 9 I. B. Rashkov, I. M. Panayotov and V. C. Shishkova, *Carbon*, 1979, 17, 103–108.

- 10 F. Pertlik, Monatsh. Chem., 1988, 119, 451-456.
- 11 M. Edstrand and G. Blomqvist, Ark. Kemi, 1955, 8, 245-256.
- 12 F. Pertlik, J. Solid State Chem., 1987, 70, 225-228.
- 13 P. A. Guńka, M. Dranka and J. Zachara, CrystEngComm, 2011, 13, 6163–6170.
- 14 R. C. Clark and J. S. Reid, *Acta Crystallogr., Sect. A: Fundam. Crystallogr.*, 1995, **51**, 887–897.
- 15 CrysAlisPro Software system, Agilent Technologies UK Ltd, Oxford, UK, 2011.
- 16 APEX-II, Bruker AXS Inc., Madison, Wisconsin, USA, 2012.
- 17 G. M. Sheldrick, Acta Crystallogr., Sect. A: Fundam. Crystallogr., 2008, 64, 112–122.
- 18 O. V. Dolomanov, L. J. Bourhis, R. J. Gildea, J. A. K. Howard and H. Puschmann, J. Appl. Crystallogr., 2009, 42, 339–341.
- 19 Diamond ver. 3.2i, Crystal Impact GbR, Bonn, Germany, 1997.
- 20 Persistence of Vision (TM) Raytracer (POV-RAY), Persistence of Vision Pty Ltd, Williamstown, Victoria, Australia, 2010.
- 21 R. Dovesi, V. R. Saunders, C. Roetti, R. Orlando, C. M. Zicovich-Wilson, B. Pascale, B. Civalleri, K. Doll, N. M. Harrison, I. J. Bush, P. D'Arco and M. Llunell, CRYSTAL09, University of Torino, Torino, 2009.
- 22 R. Dovesi, R. Orlando, B. Civalleri, C. Roetti, V. R. Saunders and C. M. Zicovich-Wilson, *Z. Kristallogr.*, 2009, **220**, 571.
- 23 F. Pascale, C. M. Zicovich-Wilson, F. López Gejo, B. Civalleri, R. Orlando and R. Dovesi, *J. Comput. Chem.*, 2004, 25, 888–897.
- 24 P. Hohenberg and W. Kohn, *Phys. Rev.*, 1964, **136**, B864–B871.
- 25 W. Kohn and L. J. Sham, Phys. Rev., 1965, 140, A1133–A1138.
- 26 A. D. Becke, J. Chem. Phys., 1993, 98, 5648-5652.
- 27 M. F. Peintinger, D. V. Oliveira and T. Bredow, *J. Comput. Chem.*, 2013, 34, 451–459.
- 28 S. Grimme, J. Comput. Chem., 2006, 27, 1787-1799.
- 29 B. Civalleri, C. M. Zicovich-Wilson, L. Valenzano and P. Ugliengo, *CrystEngComm*, 2008, **10**, 405–410.
- 30 C. G. Broyden, Math. Comput., 1965, 19, 577-593.
- 31 D. D. Johnson, *Phys. Rev. B: Condens. Matter*, 1988, **38**, 12807–12813.
- 32 P. Canepa, R. M. Hanson, P. Ugliengo and M. Alfredsson, J. Appl. Crystallogr., 2011, 44, 225–229.
- 33 A. Schulz and A. Villinger, *Inorg. Chem.*, 2009, 48, 7359–7367.
- 34 A. Schulz and A. Villinger, *Chem. Eur. J.*, 2012, **18**, 2902–2911.
- 35 R. Haiges, M. Rahm, D. A. Dixon, E. B. Garner and K. O. Christe, *Inorg. Chem.*, 2012, 51, 1127–1141.
- 36 R. Haiges, M. Rahm and K. O. Christe, *Inorg. Chem.*, 2013, 52, 402–414.
- 37 A. Simon and O. Reckeweg, Z. Naturforsch., B: Chem. Sci., 2003, 58b, 1097–1104.
- 38 E. D. Stevens, Acta Crystallogr., Sect. A: Cryst. Phys., Diffr., Theor. Gen. Cryst., 1977, 33, 580–584.
- 39 E. Garman, Acta Crystallogr., Sect. D: Biol. Crystallogr., 2010, 66, 339–351.

40 F. Corà, M. Alfredsson, G. Mallia, D. Middlemiss, W. Mackrodt, R. Dovesi and R. Orlando, in *Principles and Applications of Density Functional Theory in Inorganic Chemistry II*, Springer, Berlin, Heidelberg, 2004, vol. 113, pp. 171–232.

- 41 C. M. Zicovich-Wilson, F. Pascale, C. Roetti, V. R. Saunders, R. Orlando and R. Dovesi, *J. Comput. Chem.*, 2004, 25, 1873–1881.
- 42 S. F. Boys and F. Bernardi, Mol. Phys., 1970, 19, 553-566.
- 43 J. G. Brandenburg, M. Alessio, B. Civalleri, M. F. Peintinger, T. Bredow and S. Grimme, *J. Phys. Chem. A*, 2013, **117**, 9282–9292.
- 44 E. Madelung, Phys. Z., 1918, 19, 524.

**Dalton Transactions** 

- 45 R. Hoppe, Angew. Chem., Int. Ed. Engl., 1966, 5, 95-106.
- 46 R. Hoppe, Adv. Fluorine Chem., 1970, 6, 387.
- 47 R. Hoppe, Z. Naturforsch., A: Phys. Sci., 1995, 50, 555.
- 48 L. Glasser, Inorg. Chem., 2012, 51, 2420-2424.
- 49 E. I. Izgorodina, U. L. Bernard, P. M. Dean, J. M. Pringle and D. R. MacFarlane, *Cryst. Growth Des.*, 2009, 9, 4834–4839.
- 50 The CSD (version 5.34 with updates, May 2013) search was restricted to complexes containing  $K^+$  cation with at least one 18-crown-6 ether coordinated to it. Only structures with R factor lower than 0.05 and containing neither errors nor disorder were considered. The resulting data set comprises 554 entries.