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TUTORIAL REVIEW  
Johan A. Martens *et al.*  
Monolithic cells for solar fuels

## Monolithic cells for solar fuels†

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Hybrid energy generation models based on a variety of alternative energy supply technologies are considered the best way to cope with the depletion of fossil energy resources and to limit global warming. One of the currently missing technologies is the mimic of natural photosynthesis to convert carbon dioxide and water into chemical fuel using sunlight. This idea has been around for decades, but artificial photosynthesis of organic molecules is still far away from providing real-world solutions. The scientific challenge is to perform in an efficient way the multi-electron transfer reactions of water oxidation and carbon dioxide reduction using holes and single electrons generated in an illuminated semiconductor. In this tutorial review the design of photoelectrochemical (PEC) cells that combine solar water oxidation and CO<sub>2</sub> reduction is discussed. In such PEC cells simultaneous transport and efficient use of light, electrons, protons and molecules has to be managed. It is explained how efficiency can be gained by compartmentalisation of the water oxidation and CO<sub>2</sub> reduction processes by proton exchange membranes, and monolithic concepts of artificial leaves and solar membranes are presented. Besides transferring protons from the anode to the cathode compartment the membrane serves as a molecular barrier material to prevent cross-over of oxygen and fuel molecules. Innovative nano-organized multimaterials will be needed to realise practical artificial photosynthesis devices. This review provides an overview of synthesis techniques which could be used to realise monolithic multifunctional membrane-electrode assemblies, such as Layer-by-Layer (LbL) deposition, Atomic Layer Deposition (ALD), and porous silicon (porSi) engineering. Advances in modelling approaches, electrochemical techniques and *in situ* spectroscopies to characterise overall PEC cell performance are discussed.

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## Key learning points

- (1) The management of light, reagents, protons, electrons and reaction products in photoelectrochemical cells.
- (2) The functions and interaction of semiconductors and co-catalysts.
- (3) The most promising cell concepts existing today, and potential designs for the future.
- (4) Methods to synthesise cell components.
- (5) Methods to characterise photoelectrochemical cells and their components.

## Introduction

The reduction of CO<sub>2</sub> emissions and the quest for sustainable energy are top priorities on the world's strategic research agenda.

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† Electronic supplementary information (ESI) available: Additional details on record STF efficiencies of Fig. 1, slideshow. See DOI: 10.1039/c3cs60424a

Sunlight is an abundant renewable energy source. About 10<sup>5</sup> TW is delivered to the Earth's surface and 1.2 hours of sunlight correspond to the global energy consumption for a whole year. Drawbacks of sunlight are its diffuse and discontinuous nature necessitating its conversion into a more convenient form of energy. Photovoltaics (PV) is an established method to convert solar energy into electricity. Commercial PV with silicon single-junction cells currently are operated at 12–20% efficiency. Theoretical efficiencies for multi-band gap PV exceed 40% and organic PV shows great potential. For storage, however, electricity is a less convenient energy carrier. Storing energy in chemical bonds of fuel molecules is a more practical option and hydrogen is a candidate. PV systems and water electrolysis have been coupled to produce hydrogen and reach up to 18% efficiency,<sup>1</sup> but the



costly large platinum electrodes and peripherals together with the competitive price of hydrogen from conventional sources are obstacles to widespread use.<sup>2</sup> The cost of renewable hydrogen in principle can be reduced by integrating light-harvesting and water splitting materials in a single system. The idea for such a water photoelectrolysis device is inspired by Fujishima and Honda's discovery of the water splitting properties of titanium dioxide (TiO<sub>2</sub>) in 1972.<sup>3</sup> Incident photons on semiconductor materials such as TiO<sub>2</sub> generate electron-hole pairs. The holes abstract electrons from water molecules and protons are reduced to hydrogen using the photogenerated electrons. Subsequent investigations led to several strategies for solar hydrogen formation using semiconductors (Fig. 1). Distinction can be made between *photocatalytic* and *photoelectrochemical* (PEC) systems. In the former, multiple suspended semiconductor nanoparticles each act as a reaction centre. In a photoelectrochemical cell (PEC cell) a semiconductor photoelectrode is submerged in

aqueous solution and electrically connected to a second electrode. Initial efforts have been focused both on photocatalytic and photoelectrochemical approaches. Efficiency has been enhanced by using set-ups that are internally biased by a PV junction that is shielded from the solution, the so-called *buried* PV. In some cases, PV is connected with a photoelectrode to achieve a hybrid system (PV-PEC). While these PV-integrated systems achieved record efficiencies, the materials they are made of are prohibitively expensive. Therefore, in the last decade attention shifted to the realisation of devices based on earth-abundant materials that are both efficient and cost-effective.

Compared to solar hydrogen production, photoelectrochemical CO<sub>2</sub> reduction is an even bigger challenge. Artificial photosynthesis of carbon-based fuel has advantages such as the ease of storage, especially when the products are liquid and have high volumetric energy density. One of the earliest reports of photocatalytic CO<sub>2</sub> reduction was by Somorjai and



**From left to right: Jan Rongé, Johan A. Martens, Francis Taulelle and Tom Bosserez**

*tion of functional porous materials and their application in (photo)catalysis, adsorption and controlled release. Current research topics are related to environmental catalysis, renewable and solar fuels, pharmaceutical formulation, functional biocompatible coatings for implants, and molecular sensing.*

*Francis Taulelle started his research work at University P. and M. Curie, Paris. He investigated ionic liquids for battery applications. By using NMR, liquid and solid structures as well as interfaces can be studied without the constraint of periodicity. His two main themes currently are crystal formation by in situ NMR and NMR crystallography applied to nanoporous materials, for which he opened the field. He is involved in application of NMR to nanodevices. Director of research at CNRS, at the University of Versailles Saint Quentin en Yvelines, he is also currently guest Professor at the KU Leuven.*

*Tom Bosserez obtained his BSc and MSc in Bioscience Engineering (Catalytic Technology) at KU Leuven (Belgium), in 2012. For 1 year he worked as a project engineer for an international water purification company. In 2013 he started his PhD at the Centre for Surface Chemistry and Catalysis with Prof. Johan Martens. His research concerns the development and characterisation of photoelectrochemical cells for the production of solar fuels.*

*Jan Rongé obtained his BSc and MSc in Bioscience Engineering (Catalytic Technology) at KU Leuven (Belgium). In 2012, he started his PhD at the Centre for Surface Chemistry and Catalysis under the supervision of Prof. Johan Martens. He received a fellowship from the Research Foundation – Flanders (FWO). His research includes the development of nano-organised assemblies for photocatalysis and for solar fuel production in photoelectrochemical cells.*

*Johan A. Martens is a full professor in Bioscience Engineering at Centre for Surface Chemistry and Catalysis, KU Leuven (Belgium).*

*His core expertise is in synthesis, modification and characterisa-*



**From left to right: Gero Decher and David Martel**

*Gero Decher obtained his PhD in Organic Chemistry from the Johannes Gutenberg-Universität in Mainz. He was appointed full professor of Chemistry in 1995 at Université Louis Pasteur and distinguished professor in 2006. Since 2007 he has been a member of the Institut Universitaire de France. His main fields of interest cover the development of the layer-by-layer assembly method, (bio)functional nanoparticles, thin film devices and nano-organised multimaterials in general.*

*David Martel received his PhD in Physical Chemistry from the University of Bordeaux. He has been a researcher in the team of G. Decher since 2013. His interests comprise the structure of self-assembled materials and the photo-generation of energy.*



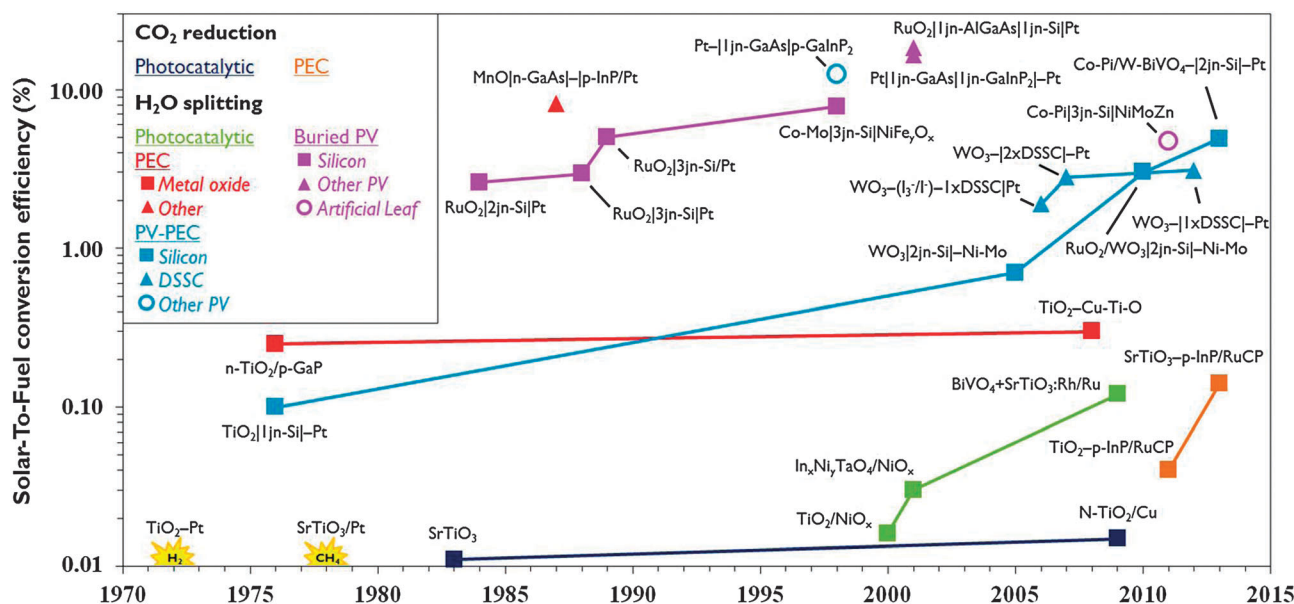
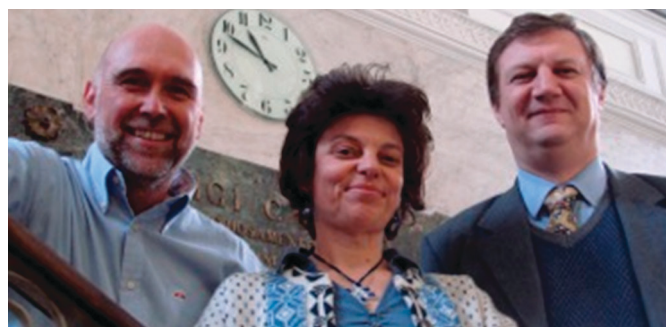


Fig. 1 Evolution of record solar-to-fuel efficiencies of different approaches, reported in the absence of chemical or electrical bias and under (simulated) solar illumination (for additional details, see ESI†).

co-workers in 1978.<sup>4</sup> Production of methane was observed when illuminating strontium titanate semiconductor coated with platinum catalyst with UV light in the presence of water and CO<sub>2</sub>. The same year, Halman reported formation of formic acid, formaldehyde and methanol on a p-type gallium phosphide photoelectrode.<sup>5</sup> While PEC systems for solar hydrogen developed rapidly, the development of systems to produce carbon-based solar fuel progresses more slowly (Fig. 1). Solar-To-Fuel (STF) efficiencies are yet to exceed 0.2%, which is the solar conversion efficiency of switchgrass, a crop often quoted for biomass production.<sup>6</sup> PV reached the

market a few decades ago and water splitting PEC cells for solar hydrogen production may soon be ready for commercial development. If solar CO<sub>2</sub> reduction is to become a viable renewable energy technology, researchers will have to join efforts in an interdisciplinary pursuit of smart reactor and electrode design.

In part A of this review critical aspects of PEC cells for CO<sub>2</sub> reduction are discussed, with emphasis on monolithic and membrane-based designs as these are the most promising. Given the parallel with solar hydrogen production, common challenges are discussed, and illustrated with remarkable



From left to right: Luca Boarino, Silvia Bordiga and Carlo Nervi

activity is mainly devoted to the characterisation of the physical-chemical properties of high surface area nanostructured materials (e.g. oxides, zeolites and MOFs) used as heterogeneous catalysts, photocatalysts, materials for adsorption, separation and storage, through *in situ* spectroscopic studies. She is the director of the Centre NIS (Nanostructured Surfaces and Interfaces [www.nis.unito.it](http://www.nis.unito.it)). She is the author of more than 300 papers listed in the ISI database (*h-index* 70).

Carlo Nervi is Associate Prof. at the Department of Chemistry of the University of Torino. He is interested in electrochemistry and DFT calculations of inorganic and organometallic complexes and their photophysical properties, with particular interest towards applications in electrocatalysis and energy conversion and storage. He has published more than 80 peer-reviewed papers in international academic journals.

Luca Boarino took a degree in Physics in 1987 at University of Torino. He joined IENG (now INRiM), working on photothermal spectroscopy in semiconductors. From 1995 he started working on porous silicon. He coordinated projects with Italian firms on porous silicon micromachining and the project ESA CHECS, monitoring the health conditions of plants on long term space missions. In the same period he observed and studied the effects of gas doping in silicon nanostructures. Since 2009 he has been the founder and responsible for Nanofacility Piemonte INRiM. His current interests are related to self-assembly, he is the author of more than 100 publications.

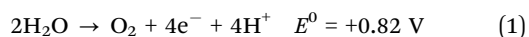
Silvia Bordiga is Associate Prof. in Physical Chemistry at the Department of Chemistry of the University of Torino. Her scientific

examples from recent literature. Based on the elaborated PEC cell requirements, in part B some upcoming technologies to conceive, build and characterise future electrode assemblies and PEC cells are presented. This review is restricted to semiconductor-based systems. The reader is referred to other reviews specifically dealing with homogeneous catalytic,<sup>7,8</sup> electrochemical,<sup>8,9</sup> biological<sup>10</sup> and thermochemical<sup>11</sup> strategies.

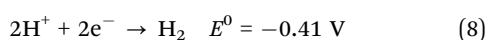
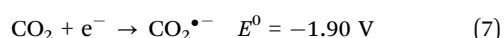
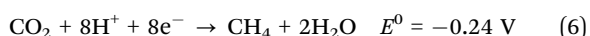
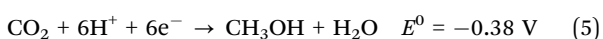
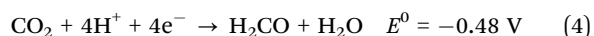
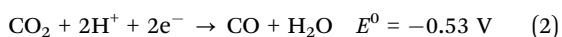
## A. State-of-the-art of photoelectrochemical cells for artificial photosynthesis

### 1. Working principle of a PEC cell

In literature, the term PEC cell is not always consistently used. In this review, to avoid confusion, *PEC cell* refers to the complete reactor containing the electrode assembly and having in- and outlets for feed and reaction products, respectively. *Electrode assembly* refers to the heart of the PEC cell, which is the solid assembly of (photo)electrodes, co-catalysts, membrane and possibly PV. PEC cells rely on the operation of a light-sensitive material that converts incident photons into electron-hole pairs, which are consumed in oxidation-reduction reactions for production of chemical fuel. Inorganic semiconductors are currently the most promising light harvesting materials because many of them possess high stability and excellent quantum efficiency. Other photosensitive materials include dyes,<sup>7</sup> quantum dots<sup>12</sup> and noble metal nanoparticles,<sup>12</sup> which may be used on their own or as sensitizers for a semiconductor. Photogenerated holes reaching the semiconductor surface can abstract electrons from water molecules that are oxidised to form molecular oxygen and protons. At pH 7 in aqueous solution the standard Nernst potential ( $E^0$ ) vs. Normal Hydrogen Electrode (NHE) at 25 °C and 1 atm gas pressure corresponds to:<sup>13</sup>



The photogenerated electrons and the protons generated according to eqn (1) can serve the reduction of  $\text{CO}_2$  to  $\text{C}_1$  molecules, such as carbon monoxide (CO), formic acid, formaldehyde, methanol and methane proceeding all through multielectron, multiproton reactions with the following standard potentials (vs. NHE, at pH 7 in aqueous solution, 25 °C, 1 atm gas pressure, 1 M for solutes):<sup>8,13</sup>



Even though  $\text{CO}_2$  reduction and water oxidation could in theory be achieved at a potential difference of 1.06 V (eqn (1) and (6)), in practice the required potential difference is much higher due

to overpotentials. The overpotential of an electrochemical reaction is defined as the difference between the potential needed to perform the reaction at a certain rate and the thermodynamically required minimum potential. In  $\text{CO}_2$  reduction reactions, overpotentials are mainly caused by the activation energy needed for electron transfer to the  $\text{CO}_2$  molecule, ohmic (electrical) losses and mass transport losses.<sup>7</sup> Among the solar fuels, formic acid and CO (eqn (2) and (3)) are most easily formed since these reactions involve only two electrons. The possibility of a methanol based economy has been explored and artificial photosynthesis of synthetic methane is an attractive route for  $\text{CO}_2$  mitigation as well. The reduction of  $\text{CO}_2$  to each of these candidate fuel molecules involves transfer of two or more electrons, making these reactions mechanistically complex and slow. Advanced catalysts are needed to accelerate and selectivate the  $\text{CO}_2$  reduction reaction. Direct reduction of  $\text{CO}_2$  (eqn (7)) is no option as it requires a very negative potential of  $-1.90 \text{ V}$  vs. NHE because no protons are involved in stabilising the carbon dioxide anion radical. Hydrogen formation occurs at a potential similar to  $\text{C}_1$  organics formation (eqn (8)) but generally is an unwanted side reaction. As a consequence,  $\text{CO}_2$  reduction catalysts are sought that have high overpotential for hydrogen formation.

The majority of scientific investigations of  $\text{CO}_2$  reduction are conducted using homogeneous catalysts dissolved in aqueous phase<sup>7</sup> and using semiconductor particles in vapour phase.<sup>14</sup> To reduce recombination losses and back reaction a more sophisticated setup is needed with spatial separation of water oxidation and  $\text{CO}_2$  reduction reactions on separate electrode surfaces. The most productive solar  $\text{CO}_2$  reduction system to date is such a PEC cell with separated electrodes.<sup>6</sup> While PEC cell performance has significantly increased over the last decade especially by improvements at the electrode level, reactor design itself received relatively little attention.

A PEC cell typically hosts two electrodes on which the water oxidation and  $\text{CO}_2$  reduction reactions take place. In the example of Fig. 2, both electrodes are photoactive and the PEC cell is divided into two compartments connected by a proton exchange membrane. PEC cells are multifunctional devices and need to control simultaneous transport of several species:

- (1) Electrons (*electronic transport*)
- (2) Protons (*ionic transport*)
- (3) Molecules (*molecular transport & barrier function*)
- (4) Photons (*optical transparency*)

**Electronic transport.** Achieving efficient electrical conduction between the anode and the cathode is essential. Electron transport can be achieved either externally, *i.e.* through a wire, or internally, by direct contact of the electrodes. When the electrodes are externally connected, current-collecting back contacts are needed. Ideally, materials with metallic conductivity such as carbon<sup>14</sup> are used, but transparent conducting oxides (TCO)<sup>15</sup> or conducting polymers such as poly(ethylene dioxythiophene)-poly(styrenesulfonate) (PEDOT-PSS)<sup>16</sup> are often chosen for their transparency.

The condition of electronic connection between the semiconductor and the catalytic sites responsible for oxidation and





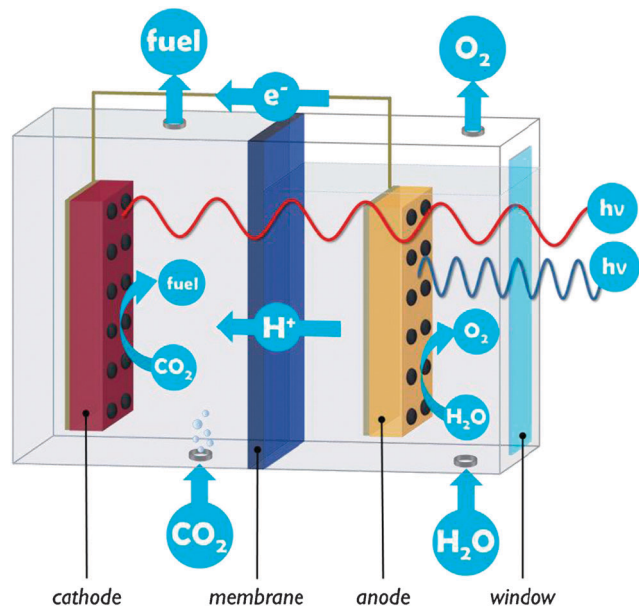


Fig. 2 Conceptual drawing of a liquid-phase, two-compartment PEC cell with photoanode, photocathode and proton exchange membrane. The pathways of light, reagents, protons, electrons and reaction products are indicated.

reduction reactions imposes high demands on the quality and continuity of the electrode materials. Materials that are active in photocatalysis may be less suited for application in a PEC cell, due to insufficient electrical conduction. Semiconductor nanostructures have to be synthesised with great care for crystal quality as defects and grain boundaries have a great influence on their performance. These can lead to excessive charge recombination<sup>13</sup> but when carefully controlled, defects may be used as a doping strategy and increase light absorption.<sup>9,17</sup> The electronic connection between the semiconductor and the catalytic centre is another critical aspect.<sup>7</sup> In this respect, the use of insulating support materials such as mesoporous silica developed by Frei and co-workers<sup>9,12</sup> is not obvious.

**Ionic transport.** In photocatalytic systems comprising two kinds of suspended semiconductor nanoparticles, negative and positive charges are transported by ions. Electrons are carried by an ionic redox mediator or redox shuttle, such as the  $\text{IO}_3^-/\text{I}^-$  redox couple, which carries electrons from one semiconductor to the other one, without being consumed in the overall process.<sup>18</sup>

In PEC cells negative charge is conveyed by electrons, and charge balance is maintained by transport of protons from the anode to the cathode (or of hydroxide ions from the cathode to the anode, in alkaline PEC cells). Proton conduction is often more critical than electron conduction, since high ionic conductivity is more difficult to achieve. Ion conducting media have two to four orders of magnitude lower conductivity than the electron conducting components used in a PEC cell.<sup>19</sup> Low proton and electron conductivities cause ohmic potential losses which result in lower overall efficiency.

Proton transport can occur either in the liquid or in the solid state. In a liquid electrolyte, operation at low or high pH is an

obvious way to increase conductivity. When changing the acidity of the medium, the pH-dependency of energy levels of redox couples, semiconductors and catalytic centres needs to be kept in mind. The operation of a PEC cell under near-neutral pH conditions enables the use of acid- or base-sensitive materials. A buffered neutral solution is least aggressive, but poses greater challenges with respect to ion transport, because proton and hydroxide concentrations are very low (around  $10^{-7}$  M).<sup>20</sup> Here, concentration gradients due to depletion of the buffer at one electrode and accumulation at the other electrode cause a concentration overpotential.

Proton transport distance is an important aspect of PEC cell design. This can be appreciated by comparing monolithic and wired variants (Fig. 3). Nocera and co-workers built a monolithic device for solar hydrogen production with 2.5% STF efficiency.<sup>20</sup> In this design, protons generated at the front side have to move around the electrode assembly to reach the back side. A wired PEC with facing electrodes reached a higher efficiency of 4.7% due to shorter proton transport distance and lower associated ohmic losses. Arai *et al.* built monolithic and wired devices for  $\text{CO}_2$  reduction, and observed efficiencies of 0.08% and 0.14%, respectively.<sup>6</sup> While a difference in proton transport distance may contribute, those authors ascribed the efficiency difference to a slight chemical bias due to a pH gradient in their wired PEC cell. In any case, these comparative studies revealed that monolithic assemblies could profit substantially from shortened proton transport paths to the cathode side. Such shortcuts could be achieved by implementing porosity in the electrode monolith.<sup>19,20</sup> Measures will have to be taken to avoid product cross-over through these openings, as will be discussed later on in this review.

The use of a solid electrolyte has significant advantages over liquids. Solid electrolytes can have high ionic conductivity. The proton conductivity of Nafion<sup>®</sup>, a widely used sulfonated tetrafluoroethylene based copolymer, can reach up to  $10 \text{ S m}^{-1}$ ,<sup>15</sup> comparable to diluted  $\text{H}_2\text{SO}_4$  solution.<sup>17</sup> Since no liquid bases or acids are involved, solid electrolytes pose no environmental hazard in the event of a leak. In a solid electrolyte, the only ionic species being transported is the one consumed in the second half-reaction, which is most often the proton. The negative counter charges in the membrane are fixed, so they

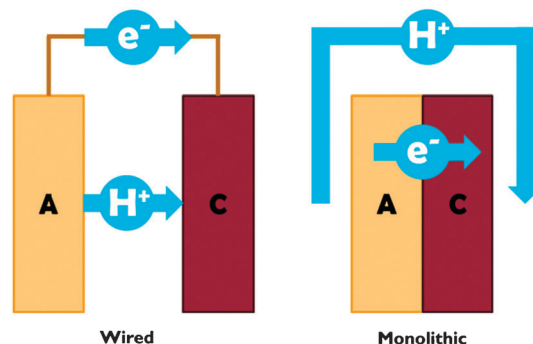


Fig. 3 Schematic representation of proton and electron transport in wired and monolithic electrode assemblies. A = anode; C = cathode.



do not generate concentration gradients and related overpotentials are avoided. Although Nafion<sup>®</sup> is regarded to be the standard in solid electrolytes, PEC cell research could benefit from new and improved materials (including anion conducting materials) with even higher conductivities while maintaining sufficient molecular barrier function.

**Molecular transport and molecular barrier function.** Molecular transport has been addressed mainly with respect to CO<sub>2</sub> availability at the catalytic sites. Since the solubility of CO<sub>2</sub> in water is very low (~30 mM at standard conditions), alternative solvents such as acetonitrile and dimethylformamide have been used, with up to eight times higher CO<sub>2</sub> solubility compared to water. Up to 1% water can be added to these solvents as a proton source without altering CO<sub>2</sub> solubility.<sup>7</sup> Pressurised and cooled PEC cells provide increased CO<sub>2</sub> partial pressures<sup>7,9</sup> and PEC cells working under supercritical conditions have been considered. Supercritical CO<sub>2</sub> is obtained at relatively mild conditions ( $T_c = 31\text{ }^\circ\text{C}$  and  $P_c = 7.4\text{ MPa}$ ) and the supercritical state provides a high CO<sub>2</sub> concentration (~20 M at 22 MPa, 35 °C).<sup>21</sup> Under these conditions, catalysts display a fivefold activity increase attributed both to enhanced catalyst stability and high reactant concentration.<sup>21</sup> To maximize these beneficial effects, the supercritical CO<sub>2</sub> should be used in a biphasic system with an additional liquid solvent containing the catalyst. Voyame *et al.* suggested the produced CO is then efficiently extracted from the liquid phase containing the catalyst due to its high solubility in the supercritical phase<sup>22</sup> (Fig. 4). Grills and Fujita proposed ionic liquids to be suitable for the liquid phase, as these have high CO<sub>2</sub> solubility and remain a separate phase even at high CO<sub>2</sub> pressure of up to tens of MPa.<sup>21</sup> In a PEC cell operated under gas or supercritical phase conditions, sufficient CO<sub>2</sub> is provided but water supply may become limiting. Also, achieving proton conductivity is less evident in the absence of liquid electrolyte. These issues may be addressed by working in a two-compartment PEC cell with a solid electrolyte separating the electrodes.<sup>23</sup>

Reaction product evacuation at either electrode can be problematic.<sup>24</sup> Water oxidation at the anode produces molecular oxygen. When the liquid flow is insufficient for diffusional-convective

transport of oxygen, gas bubble formation will introduce overpotentials. At the cathode, reaction products may have low solubility in liquid solvents and CO formation may cause gas bubble formation as well. Operation of the PEC cell in the vapour phase instead of the liquid phase can be considered. In order to prevent cathode flooding in vapour phase PEC cells, the device could be heated above the boiling point of the organic reaction products. A temperature of 100 °C will be sufficient for most of the light organic molecules formed. A cooler/condenser unit after the PEC cell enables recovery of the liquid product and recycling of unconverted CO<sub>2</sub>. In the design of a heated PEC cell, the reduced solid electrolyte conductivity at high temperatures may be a critical issue.

In single compartment PEC cells oxidation and reduction reactions occur in the same medium. The concurrent production of oxygen gas at the anode and of reduced products at the cathode dilutes the solar fuel outlet stream of the PEC cell. This entails product workup steps, and may be dangerous since flammable and even explosive mixtures (oxygen and fuel) can be formed. In addition, back reaction of the reduced products will be catalysed, which lowers product yield. For example, the popular TiO<sub>2</sub> semiconductor is a highly effective photocatalyst for oxidation of organic molecules.<sup>6</sup> In PEC cells with electrodes submerged in electrolyte solution oxygen bubbles produced from water oxidation simply rise to the surface, similar to the operation of a Hoffman apparatus or Fujishima and Honda's first PEC cell.<sup>3</sup> Since solubility of oxygen in water is very low (~1.3 mM at standard conditions), cross-over to the cathode is said to be negligible. For the gaseous reduction products CO and H<sub>2</sub>, the same strategy can be applied to avoid back reaction. Nonetheless, cross-over may result in a significant product loss of several percent in these cases.<sup>15</sup> Products such as formic acid readily dissolve and may reach the oxidation sites through diffusion in the liquid electrolyte.<sup>6</sup> The introduction of a molecular barrier function such as an ion exchange membrane is a convenient way to achieve product separation by restricting transport between anode and cathode compartments to ionic intermediates. Many membranes such as Nafion<sup>®</sup> are permeable to cations and impermeable to most molecular species. Similar to direct methanol fuel cells, there is a trade-off between enhancing molecular barrier function and decreasing proton conduction.<sup>15</sup> The use of other, porous separators (*e.g.* glass frit) has been investigated and these may be a low cost solution to limit convective reaction product cross-over.<sup>15</sup>

**Transparency.** Photoelectrochemical reactors require the input of light. As a consequence, every material located in the path between the light source and the semiconductor should be maximally transparent. This severely restricts the materials choice for a PEC cell. When both electrodes contain semiconductors, the top electrode should be transparent for longer wavelengths (lower energies) which can be absorbed by the bottom electrode (Fig. 2), *i.e.* the top electrode should always have the largest band gap. Alternatively, the photoelectrodes can be positioned side by side, but this is likely to reduce overall efficiency since the illuminated PEC cell surface area needs to be larger. The PEC cell could also be engineered to be

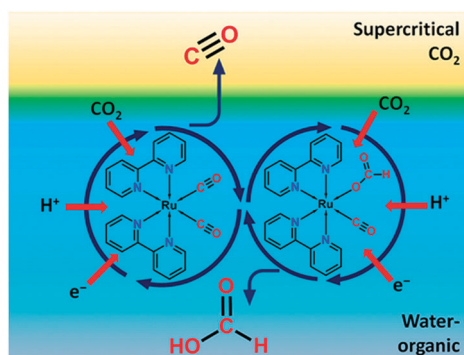


Fig. 4 Representation of a biphasic reaction medium with a supercritical CO<sub>2</sub> phase and liquid water-organic phase for homogeneous CO<sub>2</sub> reduction. Reproduced with permission from ref. 22. Copyright 2013, American Chemical Society.



illuminated from both sides by sunlight using mirrors,<sup>6</sup> although this complicates the design. Nanostructuring has proven to be an efficient strategy to reduce specular reflection and increase internal scattering for an optimal absorption of the incident light.<sup>12</sup> Co-catalysts based on *e.g.* cobalt or copper often added to photoelectrode surfaces cause parasitic light absorption. Gas bubble formation should be avoided on the illuminated electrode, because this causes light scattering and reflection losses as well.

Finally, the PEC cell casing should be transparent or contain a window to allow solar illumination. High-grade optical windows can be made of fused silica or quartz. Poly(methyl methacrylate) is a low cost alternative with good light transmission. Window materials and anti-reflective coatings that were developed for PV panels can be used. For PEC cells, there is the additional prerequisite that these materials should be chemically stable in contact with electrolyte.

## 2. PEC photosystems

Solar fuel production is initiated by the absorption of a photon in a semiconductor. The photon excites an electron from the Valence Band (VB) to the Conduction Band (CB) and a hole is left behind in the VB. The energy difference between VB and CB is called the band gap. Absorption of a photon only occurs when its energy is at least the band gap energy of the semiconductor. A small band gap material will absorb a larger part of the solar spectrum. For solar fuel production the minimum band gap is dictated by the potential energy levels of the half reactions of water oxidation and CO<sub>2</sub> reduction (Fig. 5). The CB electrons and VB holes created in the semiconductor must be at an appropriate potential for effecting these reactions.<sup>17,25</sup> Different strategies can be followed to cope with this trade-off between light absorption and reaction potentials.

Semiconductors can be used either for creating PV junctions or as photoelectrodes. A photoelectrode is a light-driven electrode that interacts with the electrolyte. Photoanodes and photocathodes perform oxidation or reduction reactions, respectively. Buried PV junctions that are formed by contact of two solid state materials and are completely shielded from the electrolyte are to be distinguished from photoelectrodes.

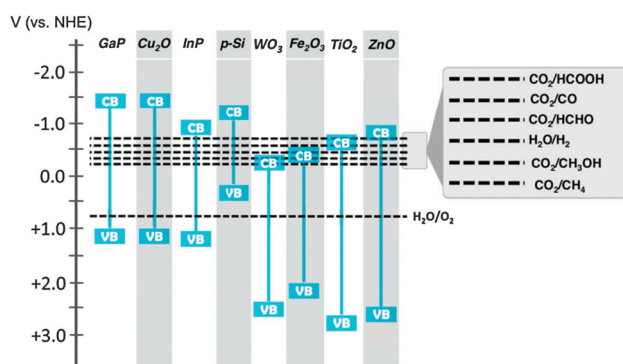


Fig. 5 Band gap energies of selected semiconductor materials and redox potentials of water oxidation and CO<sub>2</sub> reduction reactions.

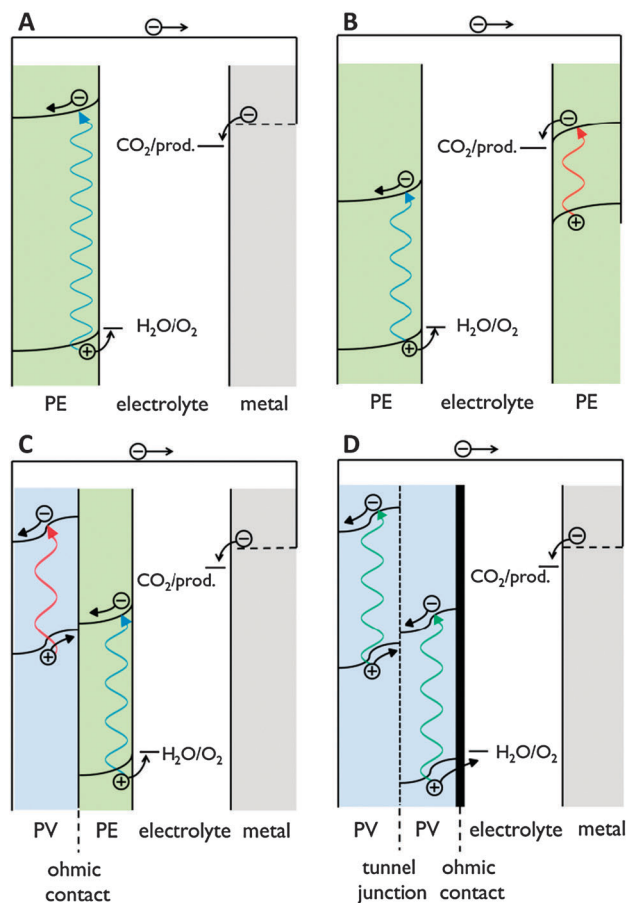


Fig. 6 Energy levels of (A) single photosystem, (B) tandem photosystem with Z scheme, (C) PV-PEC photosystem with PV junction and photoelectrode and (D) buried PV photosystem without photoelectrodes. Green: photoelectrode (PE); blue: PV junction; grey: metal cathode.

**Single photosystem.** Semiconductor materials can be used either as a photoanode or as a photocathode in combination with a metal counter electrode, mostly platinum (Fig. 6A). Performing both water oxidation and CO<sub>2</sub> reduction on a single photoelectrode can only be achieved provided the band gap is wide enough. For most materials, the usable photopotential amounts to only 50–70% of the band gap energy. Accounting for overpotentials, in practice band gap energies of  $\sim 3$  eV are required for single photosystems.<sup>17</sup> This corresponds to a photon energy in the UV part of the solar spectrum. Since sunlight contains little UV light, the use of such wide band gap materials entails a maximum theoretical solar energy conversion of only a few percent, which makes this approach less attractive for practical application.

**Tandem photosystem.** In order to make better use of the solar spectrum, two or more semiconductors that absorb in the visible range of the spectrum are needed. In Fig. 6B, the energy levels of an assembly with two photoelectrodes are illustrated. In this tandem photosystem absorption of two photons is needed to transfer one electron. Such combination of two band gaps creates a wide electrochemical window. This system is often referred to as the Z scheme and mimics solar energy



conversion in natural photosystems. The letter Z refers to the two band gaps and electron shuttling from the CB edge of the photoanode to the more positive VB edge of the photocathode. Based on arguments of band bending near semiconductor surfaces (*vide infra*), the better options for the photoanode and -cathode in Z schemes are n- and p-type, respectively. An adequate Z scheme generates VB holes with strong enough oxidation potential to extract electrons from H<sub>2</sub>O molecules, and CB electrons with a reduction power sufficient for electrochemical CO<sub>2</sub> reduction.

**Integrated PV photosystem.** A photoelectrode may receive additional bias from a PV junction. In such PV-PEC cells, one or multiple junctions are integrated (Fig. 6C). Since PV materials are sensitive to liquid electrolyte, the PV is shielded ('buried') from the reaction environment by a back contact with co-catalysts. The PV junction may even be the sole supplier of the bias. An assembly without photoelectrode containing only a shielded PV junction is referred to as *buried PV* (Fig. 6D). The PV cell may be single or multiple junction, based on Si or other elements, and dye-sensitized solar cells (DSSCs) have been used. Even though the photosystem in such buried PV cells does not contain photoelectrodes, the complete reactor is still referred to as a PEC cell, since light-driven electrochemical reactions occur in it.

PEC cells with integrated PV often have been used for solar fuel production. The highest STF efficiencies to date have been achieved by such systems. A drawback of PV based photosystems is the high material cost. A back-of-the-envelope calculation shows that a PEC cell installation producing formic acid with 2% STF efficiency and with a lifetime of 12 years should cost no more than 100 € m<sup>-2</sup> to be competitive with fossil fuel.

### 3. Semiconductor-co-catalyst nano-assemblies

Co-catalysts are added to the semiconductor surface to accelerate and selectivate the half-reactions (eqn (1)–(8)). Co-catalysts in most instances are electrocatalysts, but may be photoactive as well. The semiconductor-co-catalyst combination has to fulfil several functions: (1) light absorption; (2) charge separation and charge carrier transport, and (3) electrocatalysis. Since few single materials score well on every one of these properties, these functions are often distributed over several materials in a multi-material assembly.

**Charge separation and transport.** An absorbed photon creates an electron-hole pair in the semiconductor. In order to avoid recombination these charge carriers must be spatially separated. The key figure to consider is the charge carrier lifetime. Interfacial reactions are relatively slow and require lifetimes in the range of milliseconds, while recombination in metal oxides occurs on timescales of picoseconds to microseconds.<sup>13</sup> To improve these lifetimes, holes should be driven towards the oxidation sites and electrons to the reduction sites. This charge separation can be facilitated by creating an internal electric field in the semiconductor material.<sup>17,25</sup> In conventional PV, the electric field is realised by solid state junctions formed when two semiconductors are brought into contact (Fig. 6C and D). In PEC cells, semiconductor photoelectrodes form a semiconductor-electrolyte junction. In an n-type semiconductor,

the VB will be bent upwards near the interface, creating a space charge region (Fig. 6A). Electron-hole pairs generated in this space charge region or within hole diffusion distance of it will be efficiently separated.

In many metal oxides with a small absorption coefficient, a large penetration depth is needed to absorb enough incident light. On the other hand, these materials often suffer from a short hole diffusion length, so charge carriers would be generated too far from the space charge region. To overcome this problem, nanosizing is beneficial and several nanoscale morphologies have been reported.<sup>2,12</sup> One-dimensional nanostructures such as nanotubular electrodes orthogonalise photon and hole paths: photons penetrate the nanotubes in the longitudinal direction, while holes generated in the nanotube wall are at nanometre distance from the nanotube surface (Fig. 7). A drawback of nanostructured systems is that they often lack band bending.<sup>12</sup> In nanostructured single semiconductors, due to the small dimensions generally no internal electric field can be generated. Charge separation is solely driven by the chemical processes at the semiconductor surface. Water oxidation consumes holes and so drives electrons to the cathode. In this case charge separation is dependent on reaction kinetics.<sup>13</sup> The integration of a second semiconductor material generates a heterojunction which drives the separation of electrons and holes because of a difference in band edge energies.<sup>13</sup> Interestingly, an electric field in nanosized semiconductor can also be induced by adsorption of a surface layer of charged anions. An electrostatic field generated by a phosphate layer on TiO<sub>2</sub> nanocrystals was shown to drive away electrons from the surface and enhance charge carrier lifetime.<sup>12</sup>

When electron transport is problematic, the material may be doped with electron donors. Besides increasing the electron density of the CB, these also reduce the width of the space charge region.<sup>17</sup> Grätzel and co-workers developed host-guest structures in which a material with high electron conductivity serves as a high surface area substrate for an ultrathin layer of photoactive material.<sup>17</sup> This approach improves both electron and hole transport. WO<sub>3</sub> semiconductor scaffolds can be used

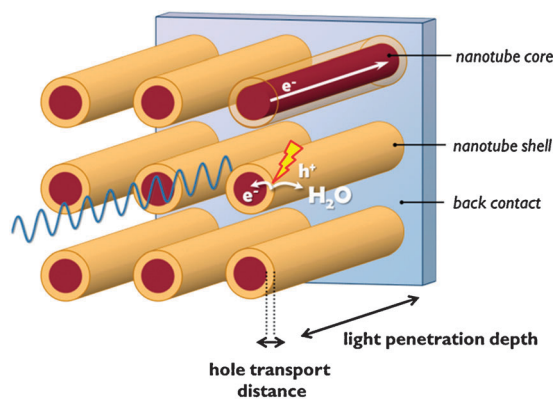


Fig. 7 Representation of light absorption and charge transport in nanotubular electrode assemblies. The nanotubes can contain a (semi)conductor core for enhanced electron conduction or light absorption, or can be hollow for increased surface area.



and carbon structures are also efficient in extracting electrons from the photoactive material.<sup>14</sup> A final and obvious strategy to reduce recombination involves the use of a catalyst that accelerates reaction kinetics and allows shorter charge carrier lifetimes.

**Electrocatalysis.** Water oxidation is the most demanding half reaction in hydrogen producing PEC cells. In carbon fuel producing PEC cells, achieving efficient CO<sub>2</sub> reduction is the most challenging task. Even when illuminated photosystems provide electrons with sufficient thermodynamic driving force for performing the reduction reactions, an inefficient electrocatalyst for these reactions causes unacceptable reaction overpotentials. CO<sub>2</sub> is the oxidation product of combustion reactions and is a very stable molecule.<sup>21</sup> The kinetic barriers for CO<sub>2</sub> activation can be relaxed by finding a catalyst that can break the linear symmetry of the CO<sub>2</sub> molecule and by favouring C–H bond formation.<sup>7</sup> The latter is known as proton-coupled electron transfer. Thus, to improve reaction kinetics, compounds that catalyse the more favourable multi-electron, multiproton reactions are needed.

The performance of a co-catalyst is commonly expressed using the following parameters:<sup>7</sup>

**Catalyst selectivity (CS).** This figure is commonly used to quantify the formation of side products.

$$CS = \frac{\text{moles of carbon atoms in intended reaction products}}{\text{moles of carbon atoms in all reaction products formed}} \quad (9)$$

**Turnover number (TON).** The TON is often reported together with the reaction time. When the TON is reported over the whole lifetime of a catalyst, it is a measure of catalyst stability.

$$TON = \frac{\text{moles of intended reaction product formed}}{\text{moles of catalytic sites}} \quad (10)$$

**Turnover frequency (TOF).** TOF is a measure of catalytic activity. One must be careful when comparing TOFs, as most catalysts have low stability and initial TOFs can be higher.

$$TOF = \frac{TON}{\text{reaction time}} \quad (11)$$

**Faradaic efficiency (FE).** FE is useful to describe electrochemical selectivity and it is an elegant way to verify the occurrence of side reactions or product cross-over.

$$FE = \frac{\text{moles of electrons consumed in reaction product formation}}{\text{total moles of electrons transferred from anode to cathode}} \quad (12)$$

CO<sub>2</sub> reduction catalysts can roughly be divided into three groups: (i) all-inorganic metal catalysts, (ii) organic molecules, (iii) metal–organic complexes. Enzymatic conversions, although inspiring, will not be discussed here.<sup>2,8</sup>

Metal co-catalysts are usually loaded on a semiconductor surface as nanoparticles. They are able to form reaction products beyond two-electron reduction, yielding sometimes even longer alkanes (C<sub>2</sub>–C<sub>9</sub>) especially in vapour phase reactions.<sup>23,26</sup> The main drawback of metal catalysts is their poor selectivity.

A single reaction product is seldom obtained.<sup>23</sup> Many metals are also prone to deactivation by poisoning.<sup>7,26</sup> Copper, being used for industrial-scale production of methanol from syngas, is an all-time favourite for CO<sub>2</sub> reduction studies.<sup>26</sup> Besides copper, noble metals (Au, Ag, Pd, Rh, Ru) are effective catalysts, and also Fe, Pb and Ni have been used.<sup>7,23</sup> Pt is less suitable for CO<sub>2</sub> reduction, as it favours H<sub>2</sub> formation.<sup>14</sup>

All-organic catalysts are metal-free and eliminate the need of expensive transition metals. A breakthrough was achieved recently by Bocarsly and co-workers who reported homogeneous CO<sub>2</sub> reduction using pyridinium compounds.<sup>27</sup> Pyridine was found to catalyse the 6-electron reduction of CO<sub>2</sub> to methanol, with formic acid and formaldehyde as intermediate products. This is remarkable, since pyridine can only transfer one electron at a time. Similar results had been obtained before using tetraalkylammonium salts and aromatic nitriles and esters.<sup>27</sup> Organic catalysts generally have high product selectivity and most of them can be operated at low overpotentials.

Metal–organic catalysts are versatile materials since both the metal centre and the organic ligands can be tuned, which results in high selectivities.<sup>7</sup> These complexes can adopt multiple redox states and can, in principle, perform multi-electron-transfer reactions. However, in almost all cases CO or formic acid is formed and products beyond 2-electron reductions are not found.<sup>21</sup> Distinction can be made between macrocyclic complexes, phosphine complexes and polypyridyl complexes. Macrocyclic complexes can be divided into tetraazamacrocycles, the most popular being Ni(cyclam), and metallomacrocycles, such as porphyrin-like materials. They mostly include Ni, Co or Fe centres.<sup>8</sup> Pd is typically used as the metal centre in phosphine compounds.<sup>8</sup> The most studied polypyridyl complex is Re(bpy)(CO)<sub>3</sub>X (bpy = 2,2′-bipyridine and X = Cl or Br). Most of these are based on Re, Ru or Os.<sup>7</sup> In general, complexes with bipyridine ligands have better product selectivity and require lower overpotentials. Currently their applicability is limited by low stability and low TOF. These issues have been partially addressed by changing the reaction medium (see Section A.1). Improved performance can also be expected through engineering of the ligands, but a deeper mechanistic understanding will be required for such directed optimization.<sup>8</sup>

For application in PEC cells, the linkage between the co-catalyst and the semiconductor surface is crucial. Catalyst anchoring can be achieved by direct covalent bonding or as a polymeric film.<sup>6,7</sup> Besides improved electronic transport, immobilisation of catalysts leads to enhanced stability and allows the use of materials that are insoluble in the electrolyte phase.<sup>7</sup> Covalent attachment is achieved by using ligands with reactive groups, such as carboxylic acid or phosphonic acid (Fig. 8). The nature of these linkers determines the quality of the electronic connection. This aspect has been investigated by Ishitani and co-workers for supramolecular homogeneous complexes.<sup>28</sup>

For the water oxidation half-reaction commonly noble metal-based oxides such as RuO<sub>2</sub> or IrO<sub>x</sub> were used with good performance.<sup>25</sup> Unfortunately, the high cost of these materials may impede their practical application. Through better understanding of water oxidation reaction mechanisms and the



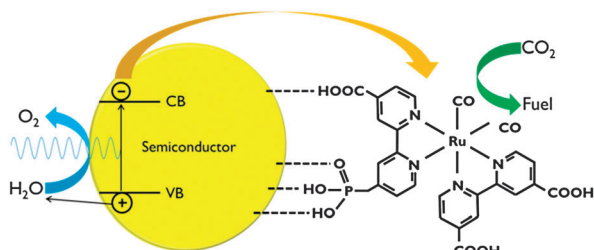


Fig. 8 Representation of  $\text{CO}_2$  reduction using a polypyridyl catalyst anchored to a semiconductor surface. In this scheme, both carboxylic and phosphonic acid linking groups are shown.

functioning of related natural enzymes, catalysts have been developed based on Mn and Co which work at low overpotentials and are less costly.<sup>2,12</sup>

**Stability of semiconductor-co-catalyst assemblies.** The solar-to-fuel reaction chain comprises light absorption, charge separation and electrocatalytic processes. Ultimately, the applicability of an assembly will be dependent on its stability. PEC cells are operated under oxidative conditions at the anode and reductive conditions at the cathode, and subjected to constant illumination. Often, corrosive liquid electrolytes will be used. Operation in the gas phase or at neutral pH in the liquid phase is expected to be less aggressive. Photocorrosion occurs when energetic electrons reduce or when holes oxidise the semiconductor material instead of the reagent, because of unfavourable band

edge positions and overpotentials.<sup>25</sup> In general, small band gap materials have less stable bonds and are more susceptible to photocorrosion. Silicon, sulphide and phosphide photoanodes are prone to photo-oxidation, while oxide photocathodes may suffer from photoreduction. Photocorrosion has been prevented by applying protective coatings or co-catalyst layers for rapid charge carrier extraction.<sup>12</sup> Co-catalyst stability can be a problem too. During multi-electron reaction cycles, the ligand environment of catalyst complexes is significantly altered. This is often associated with ligand exchanges, which makes the catalyst susceptible to inactivation. For molecular catalysts, this is especially true when operating in solvents that tend to coordinate the metal centre of the catalytic complex.<sup>21</sup> Some water oxidation catalysts such as cobalt phosphate and related compounds deal with these issues in a unique way: they self-assemble and are thus self-healing in the presence of appropriate electrolytes.<sup>20</sup>

#### 4. Monolithic assembly concepts

The preceding sections have made clear that there is an inextricable link between the development of photosystems and the design of a PEC cell in which the electrode assemblies are operated. Bearing the multitude of requirements in mind, promising concepts from literature are discussed in this section (Fig. 9).

**Artificial leaf.** The artificial leaf targets a wireless, monolithic assembly for practical generation of solar fuels, similar to the leaf of an energy crop. The appealing idea has been around

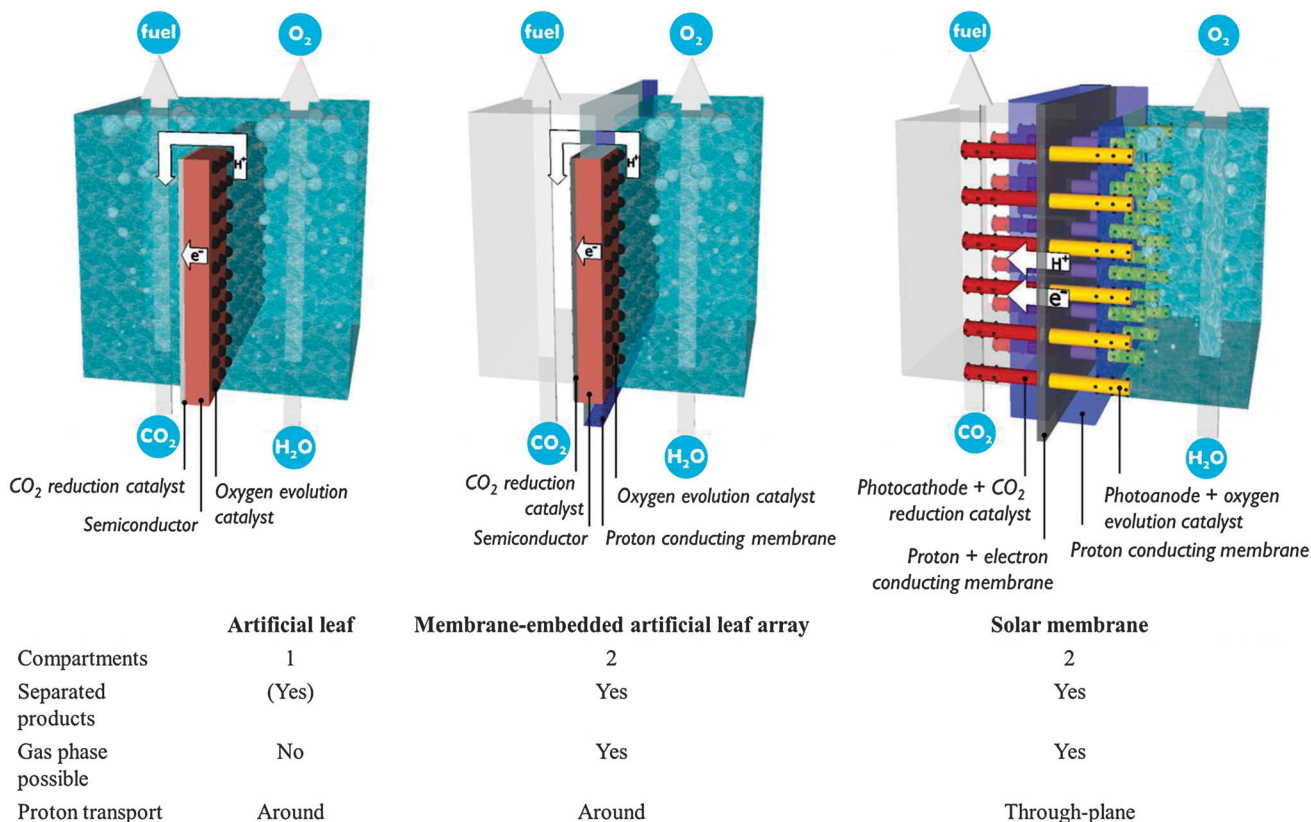


Fig. 9 Monolithic electrode assemblies and their advantages.





for some years and received renewed interest since Nocera and co-workers demonstrated an artificial leaf based on low-cost silicon PV interfaced with self-healing catalysts based on earth-abundant elements.<sup>20</sup> Their device could be operated in a neutral pH electrolyte. One of the first examples of such a stand-alone, monolithic assembly was given by Rocheleau *et al.* in 1998. They produced hydrogen at 7.8% STF efficiency using a buried triple junction silicon PV, interfaced with co-catalysts.<sup>29</sup> Morikawa and co-workers demonstrated an artificial leaf that reduces CO<sub>2</sub>.<sup>6</sup> Today, the term artificial leaf is used for monolithic assemblies carrying a wireless current, mostly operated under mild conditions and containing self-assembling co-catalysts.

The artificial leaf benefits from reduced cost and easy operation. Hydrogen production has been demonstrated by simply depositing an artificial leaf in any recipient filled with water. Even polluted water and sea water have been claimed suitable. Unfortunately, the requirement of a liquid phase reduces CO<sub>2</sub> availability. Gaseous products could be kept separated by collecting them on both sides of the planar monolith assembly. Such a separation strategy is successful when employing metre-scale panels.<sup>1</sup> However, ionic transport over such macroscopic distances imposes tremendous ohmic losses, as discussed in Section A.1. The trade-off between product separation and proton transport distance restricts the size of an artificial leaf. A computational model has shown that electrode widths of 30 mm already lead to several hundreds of mV overpotential.<sup>15</sup>

**Membrane-embedded artificial leaf array.** The anode to cathode proton transport distance can be shortened using a proton exchange membrane. Such membrane allows protons to cross but acts as a molecular barrier for reagents and reaction products. Furthermore, by compartmentalisation one or both electrodes could be operated in the gas phase. One could imagine the use of flue gas or concentrated atmospheric CO<sub>2</sub> as the input stream for the cathode side of membrane-embedded artificial leaves. The concept of arrays of membrane-embedded leaves was considered for modelling purposes.<sup>15</sup> Haussener *et al.* calculated that the size of these individual small leaves should not exceed ~10 mm to avoid excessive overpotentials.<sup>15</sup> Still, at this scale non-negligible transport losses are calculated even using 1 M H<sub>2</sub>SO<sub>4</sub> electrolyte.<sup>15</sup> To enhance performance, the size of individual leaves could be scaled down to the micrometre or even nanometre scale. Proton transport distances around nanoleaves will be short and a strong acid electrolyte will no longer be needed. A disadvantage of membrane-embedded leaves is that the membrane part of the assembly is not photoactive. Haussener *et al.* considered 10% surface area loss for the membrane function.<sup>15</sup>

**Solar membrane.** When photoactive components become extremely small and an integral part of the interface, one could speak of a solar membrane. This concept mimics the thylakoid membranes performing photosynthesis in natural leaves. The solar membrane concept could be elaborated along two lines of research: membrane-embedded silicon microwires and fuel cell-like PEC cells comprising porous gas diffusion electrodes,

coated with a photocatalyst. The Lewis group at Caltech is exploring a multifunctional interface comprising silicon microwires that are decorated with a co-catalyst.<sup>16</sup> The polymer serves a multitude of functions. It serves as a structural support for the silicon microwires, allows ion transport while avoiding product mixing, provides electronic contact between the anode and the cathode and is transparent for long wavelength photons. These demands were met by embedding silicon microwires in a Nafion<sup>®</sup> proton exchange polymer. Two such electrodes were connected by a thin PEDOT-PSS layer. This layer serves as an electrical interconnect while still providing sufficient proton conductivity. The membrane-embedded microwires illustrate that the combination of different materials at appropriate dimensions is a successful strategy for obtaining an assembly that fulfils multiple functions. Unfortunately, a complete, working PEC cell built according to this principle has not yet been demonstrated.

A second approach builds upon the experience with fuel cells and imposes less stringent requirements on the interface and the photoelectrode. In these PEC fuel cells, a membrane-electrode assembly is prepared in a way similar to a fuel cell assembly.<sup>23,24,30</sup> In a typical assembly, carbon-based gas diffusion electrodes are coated with semiconductor nanoparticles or, alternatively, a titanium metal sheet is anodized to obtain a porous TiO<sub>2</sub> film. A counter electrode is operated in the dark and typically consists of metal nanoparticles deposited on a carbon substrate. These components are hot-pressed to obtain a monolithic membrane-electrode assembly and inserted in a two-compartment reactor. In 1996, Ichikawa and Doi produced methane in a PEC fuel cell using TiO<sub>2</sub> thin film as the photoanode and a ZnO/Cu co-catalyst as the cathode.<sup>30</sup> Centi and co-workers reported formation of long-chain alkanes by operating the cathode compartment in the gas phase.<sup>23</sup> They employed Fe co-catalyst on carbon nanotube supports in combination with a TiO<sub>2</sub> nanotube photoanode fabricated by anodisation. In these PEC fuel cells only one side is illuminated, which allows non-transparent electrode materials to be used. Electrical connection is achieved by an external wire, so common proton exchange membranes may serve for the interface. However, reported efficiencies are still quite low. Product evacuation and proton transport are suboptimal in the current embodiment.<sup>24</sup> Current PEC fuel cells contain a single photosystem. Performance is expected to improve by integration of multiple small band gap semiconductors, which will drive charge separation and enhance visible light absorption.

In contrast to the artificial leaf, in a solar membrane, the efficiency will not decrease when the PEC cell surface area is increased. The porous electrodes allow short proton conduction pathways, which is important for operation at mild pH, or in vapour phase. The solar membrane concept offers assemblies with good light harvesting capabilities and the possibility to fine tune each side individually. At the same time, the membrane is a weakness of these devices. Besides the high cost, a proton exchange membrane is sensitive to poisoning by alkali and alkaline earth cations present in most water sources. When operated using water in the vapour phase, this problem



is largely avoided. Care must be taken to avoid dehydration of the membrane, however, because proton conductivity is critically dependent on hydration. Clearly, there is a strong need for research on these multifunctional interfaces.

Even though promising concepts have already been demonstrated, none of the current PEC cells fulfil all four requirements for practical application: (i) high STF efficiency, (ii) low cost, (iii) high durability and (iv) suitability for scale-up. While research focus in the past was mostly on the first three conditions, suitability for scale-up should also be considered early on in the design of assemblies and PEC cells. This implies that on an industrial scale, the synthesis techniques should be accessible at a reasonable cost and the reaction products should be safely collected. The cost of industrial-scale solar fuel plants has already been addressed by Pinaud *et al.*<sup>1</sup> Through a cost analysis, it has been suggested that integrated PEC devices could possibly produce hydrogen at a lower cost than a PV-coupled electrolyzer system.<sup>1</sup> Finally, since the purpose of solar fuel research is to provide a sustainable alternative to fossil fuels, the environmental impact of PEC cell production and operation is an essential factor when evaluating different approaches. This was studied by Zhai *et al.* using life cycle assessment.<sup>31</sup>

## B. Emerging tools for artificial photosynthesis

### 1. PEC cell component synthesis techniques

PEC cells are built for optimal transport of light, molecules, protons and electrons. As such, composite multimaterials have to be brought together in a compatible assembly with careful attention for all functions. The synthesis method determines to a large extent the quality and performance of an electrode assembly. Such a multifunctional system puts strong demands on the precision of the technique, yet the targeted broad application of solar fuel production prescribes low-cost manufacturing. The ideal technique allows for large-scale production yet nanoscale control over the final assembly. Instead of attempting an exhaustive overview of electrode synthesis techniques, here a selection of techniques is discussed that meets these requirements.

**Silicon engineering.** Almost all of the high-efficiency set-ups (Fig. 1) involve PV junctions to provide the thermodynamic driving force, so it seems likely that any successful solar fuel device will adopt the same strategy to benefit from the advancements made in the field of PV. PV is based on internal junctions and avoids the stringent requirements associated with direct semiconductor–liquid junctions,<sup>29</sup> so it effectively decouples photovoltaic and electrochemical functions in a PEC cell. The most realistic PV material is silicon because of its relatively low cost. Silicon is the second most abundant element in the Earth's crust and its use in PV led to commercial success. It has already attracted appreciable attention in PEC cell research, both in buried PV devices and in hybrid PV-PEC cells. Since water oxidation and CO<sub>2</sub> reduction require a large potential

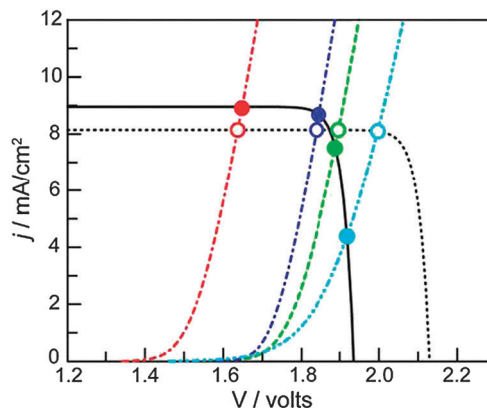


Fig. 10  $J$ - $V$  curves of a triple junction Si PV (solid black line), idealized PV (dashed black line), and different water oxidation catalysts (colored lines). The intersections indicate the working point of the assemblies. Catalysts with lower activity (light blue curve) are more strongly affected by the potential difference that can be generated by the PV (compare empty and filled light blue circles). Reproduced with permission from ref. 32. Copyright 2012, National Academy of Sciences.

difference and silicon is a small band gap material, multiple junctions are necessary. Each sub-cell should absorb a specific part of the solar spectrum, and the characteristics (short circuit current, open circuit voltage) should be optimized to coincide with the  $J$ - $V$  characteristics of the co-catalysts. The cross-section of both curves yields the operating point of the PEC cell (Fig. 10). A catalyst with low overpotential allows the use of a PV stack that delivers a potential near the thermodynamic potential for the reactions of interest. Such closely matched PV minimizes potential energy losses but is very sensitive to an increase in overpotential.<sup>32</sup>

Since dense silicon wafers impede proton transport in a PEC cell, silicon microwires are under investigation that allow through-plane transport<sup>16</sup> (Fig. 7). Such microwires also increase surface area and provide short charge carrier paths, similar to nanotubular structures. As a consequence, less pure and less costly silicon can be used.<sup>12</sup> Microstructured silicon photoelectrodes can be put in contact with the electrolyte to generate a semiconductor–liquid junction. To further increase charge carrier separation, an internal radial junction can be introduced.<sup>16</sup> A great benefit of silicon microwires, when embedded in a polymer membrane, is the flexibility. The introduction of microwires on both sides potentially affords a flexible electrode assembly with high efficiency. Silicon microwires have been tested as a photocathode but are less suitable as a photoanode. Metal oxides of similar morphology could take up this role. Alternatively, silicon photoanodes can be fabricated that are stabilised by a protective charge tunnelling layer.

The inverse morphology of silicon wires, *i.e.* porous silicon (porSi) was developed in the early '90s.<sup>33</sup> PorSi can be obtained by electrochemical etching of silicon in HF,<sup>33</sup> or by a simple electroless etching procedure in HF–H<sub>2</sub>O<sub>2</sub> solution, catalysed by metal particles.<sup>34</sup> Combined with nanopatterning of surfaces by means of nanosphere self-assembly, the latter offers new possibilities to create ordered pores with controllable size.<sup>34</sup>



PorSi is a highly tunable material. Pore morphology ranges from unidirectional pores to dendritic structures or even nanopillars at very high porosities. The material is applied in PV cells as an antireflection coating, and devices using porSi as the sole light absorber have been reported.<sup>35</sup> The application of porSi membranes in fuel cells is of particular interest for PEC cells. A silicon-based inorganic-organic membrane was fabricated with higher proton conductivity than Nafion<sup>®</sup>, no volumetric size change, and when used in a fuel cell at low humidity, it produces a power density that is an order of magnitude higher than other systems.<sup>36</sup> These advantages were obtained by creating pores with diameters of 5–7 nm, adding a self-assembled molecular monolayer on the pore surface, and capping the pores with a layer of porous silica. The limited sensitivity to low humidity is an interesting feature for PEC cells operating in the vapour phase and at ambient temperature. If porSi would be used in a PEC cell, the functions of photoelectrode and membrane would be combined in a single material, seriously reducing PEC cell cost. The band gap of porSi is widened compared with dense Si, and dependent on porosity.<sup>33</sup> This allows tuning of the band gap for optimal solar absorption and band edge positions for the requested reactions. If a porosity gradient is applied to the structure, it is speculated that the resulting internal electric field could enhance charge separation. After etching, electronic conductivity of porSi is decreased, caused by the discontinuous morphology. Free charge carrier populations can be influenced however, by interaction with adsorbed species on the large internal surface area of porSi (up to 600 m<sup>2</sup> cm<sup>-3</sup>). The adsorption of NO<sub>2</sub> or NH<sub>3</sub> can turn porSi into a p-type or n-type semiconductor, respectively.<sup>37</sup> Overall, porSi is a small band gap, high surface area material that reduces PEC cell cost by using less Si of lower purity and by replacing the membrane component. Although it has not been investigated in PEC cells until now, recent developments encourage the introduction of this versatile material.

**Layer-by-layer deposition.** It is now a little over 20 years since Layer-by-Layer (LbL) assembly was introduced as yet another method to functionalize surfaces and to fabricate thin films.<sup>38</sup> It is based on the consecutive adsorption of two or more interacting components (polyelectrolytes, proteins, DNA, biological, inorganic or organic nanoparticles) from solution which results in a multilayer structure (Fig. 11). Most LbL films are assembled using electrostatic interactions between oppositely charged materials, but many other interactions have been exploited as well. The deposition process is dependent on control parameters such as solution pH, ionic strength, deposition time, nature of the adsorbing species and others. In each adsorption step a monolayer is deposited, allowing for nano-scale control over layer properties such as film thickness and composition.

Fig. 11 shows the deposition of LbL films by dipping, but methods such as spray-assisted<sup>39</sup> or spin-assisted LbL assembly or inkjet printing<sup>40</sup> have been successfully employed as well and scaled-up roll-to-roll processes are already used in industrial production enabling the coating of millions of m<sup>2</sup> per year.

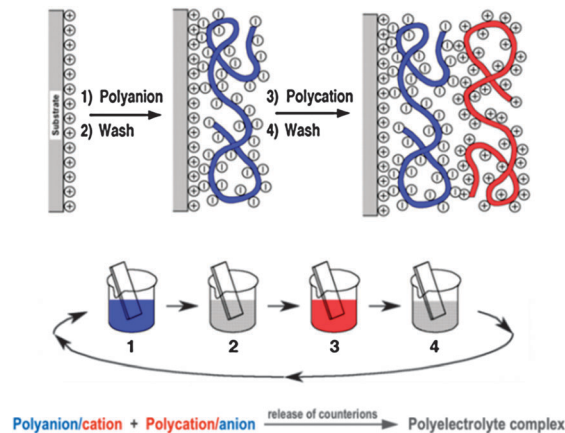


Fig. 11 Top: simplified molecular conception of the first two adsorption steps in LbL deposition. The polyanion conformation and layer interpenetration are an idealization of the surface charge reversal with each adsorption step which is the base of the electrostatically driven multilayer build-up depicted here. Counterions are omitted for clarity. Middle: schematic of the film deposition process using glass slides and beakers. Steps 1 and 3 represent the adsorption of a polyanion and polycation respectively, and steps 2 and 4 are washing steps. The four steps are the basic build-up sequence for the simplest film architecture (A/B)<sub>n</sub>, where *n* is the number of deposition cycles. Bottom: general form of the chemical equation for polyelectrolyte complexation.

Consequently LbL assembly has spread into a large number of different areas of applications (PV, optoelectronic devices, energy storage, coatings, microreactors, flexible electronics, tissue engineering, biosensors, (bio)catalysis, ...) over the last years and led to substantial or even groundbreaking advances in many cases.

LbL has not yet been applied in PEC cell research but was found useful in related fields, including photocatalysis, fuel cells and PV.<sup>38</sup> The composite nature of PEC cells makes LbL assembly an excellent method for the fabrication of cells. We believe that multi-component LbL films could find their merits in PEC cells with respect to electrode construction, band gap engineering, dye sensitisation and light harvesting, barrier properties, electrical and proton conduction, corrosion prevention, pore design or catalyst layers. LbL films have already been used in the past as examples for all of these functions so the door to finding the right components and film deposition conditions for matching PEC cell requirements is, in our opinion, wide open.

**Atomic layer deposition.** Atomic Layer Deposition (ALD) is a sequential process of self-limiting surface reactions that result in the formation of ultrathin films on an atomic level. Each cycle begins with a pulse of precursor in a reaction chamber with chemisorption of the precursor to a functional substrate until saturation of the surface (Fig. 12). Excess precursor is removed after the chemisorption step and a pulse of hydrolysing or oxidising agent is then introduced. Chemical reaction with an adsorbed precursor ensures the formation of a functional monolayer and, after evacuation of the gases, the cycle is repeated. ALD differs from chemical vapour deposition, in which the reactants are introduced simultaneously and layer





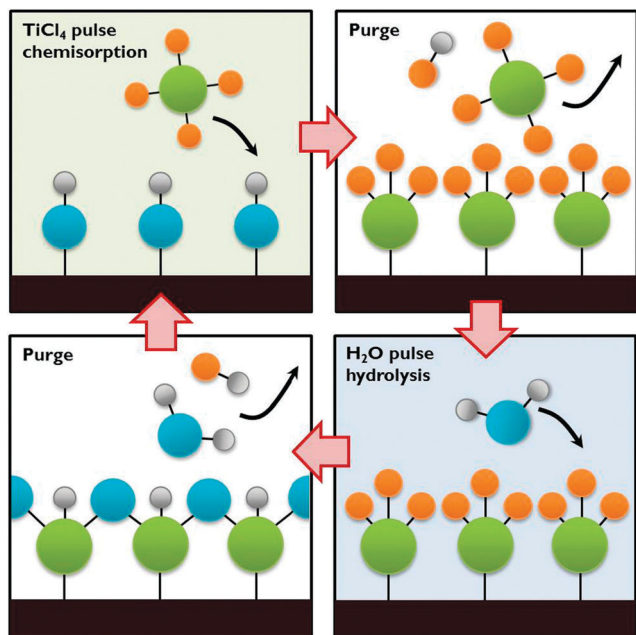


Fig. 12 The ALD process exemplified by  $\text{TiCl}_4$  chemisorption on a hydroxylated surface followed by hydrolysis.

growth rate depends on the gas flow. By changing the number of ALD cycles the thickness of the film is precisely controlled at the subnanometre scale. By performing the deposition in gas phase a conformal coating can be deposited even on complex 3D nanostructures, such as nanopores, -wires or -tubes.<sup>41</sup>

A wide choice of materials is accessible with ALD, many of which bear promise as a photoanode or photocathode in a PEC cell, especially when deposited on a nanostructured substrate such as carbon or silicon nanomaterials. Even composite thin films can be synthesised by the sequential deposition of different precursors, in order to increase charge separation and light absorption. Such nanoscale heterojunctions show great potential if the composite layer size can be precisely tuned.<sup>12,13</sup> ALD can be combined with *in situ* doping to manipulate band gap energetics and enhance conductivity. Co-catalysts such as transition metals, metal oxides and even alloys can be deposited with tuneable sizes. All these possibilities make this technique ideally suited for PEC cell development. As discussed in Section A.3 some photoelectrodes are prone to corrosion. ALD may offer a way to protect the electrodes by applying a protective thin film. This protective layer shields the material from the electrolyte and passivates surface states. ALD has been successfully used by Grätzel and co-workers for protecting an unstable p- $\text{Cu}_2\text{O}$  photocathode with several nanolaminate  $\text{ZnO}/\text{Al}_2\text{O}_3/\text{TiO}_2$  layers.<sup>12</sup> This approach is also applicable to co-catalysts and could in general increase the PEC cell performance and lifetime.<sup>42</sup>

So far ALD has been used mainly in microelectronics for the deposition of highly insulating layers of only a few nm onto small surfaces. As the scale of ALD is only limited by the size of the reaction chamber, large scale production of photoelectrodes for PEC cells seems feasible. When producing multimaterial

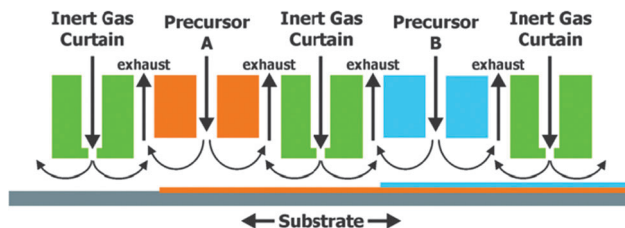


Fig. 13 Schematic example of spatial ALD with separation of different precursor reactions by an inert gas curtain. Horizontal movement of the substrate leads to the sequential deposition of monolayers. Reproduced with permission from ref. 43. Copyright 2012, American Vacuum Society.

films with layers of several hundredths of nm on larger substrates, the slow film growth (typically  $0.1 \text{ nm s}^{-1}$ ) could be impractical. To overcome this limitation alternatives such as spatial ALD have been investigated.<sup>43</sup> Spatial ALD (Fig. 13) can even be applied in roll-to-roll printing. The sequential and self-limiting nature of ALD is perfectly compatible with LbL. One could imagine a roll-to-roll printing process in which LbL deposition steps are alternated with ALD treatments.

## 2. Elaboration of future PEC cell designs

The ALD, LbL and porSi material tools could be used to create a next generation of monolithic assemblies. In this section we describe two innovative approaches to illustrate where future research could lead.

**Inorganic semiconductor solar membrane.** As described in the previous section, porSi could serve as the interface in a solar membrane assembly. In the proposed design shown in Fig. 14, pore walls are coated by ALD with a wide band gap semiconductor which acts as a photoanode. The photoanode provides protection against photo-oxidation of the silicon semiconductor and is part of the tandem photosystem. If needed, the photoanode itself can be protected from photocorrosion by applying a passivation layer using ALD. By adopting asymmetric pores, incident light will first encounter the wide band

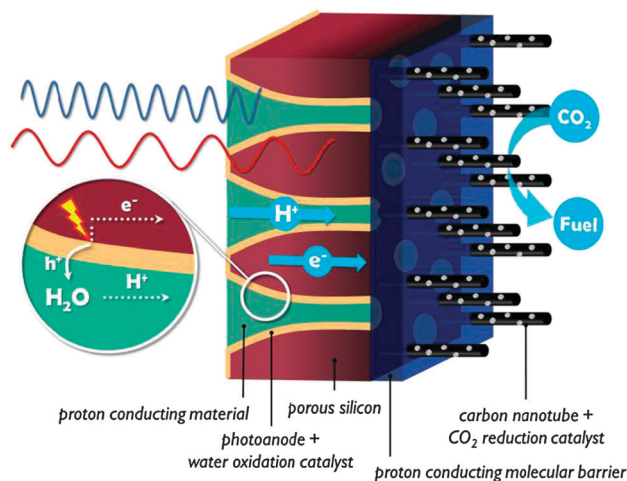


Fig. 14 Solar membrane based on porSi. The pores are decorated with photoanode material and filled with proton conducting material.



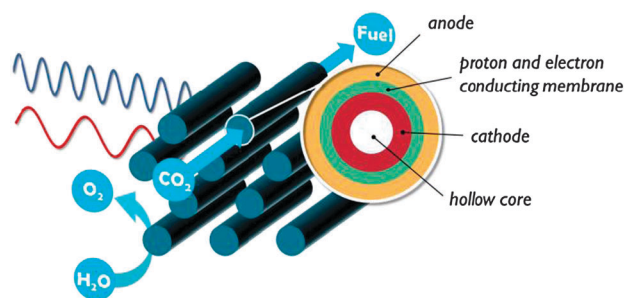


Fig. 15 Concentric multitubular assembly obtained by layered deposition of organic polymers.

gap photoanode which absorbs short wavelengths of the solar spectrum. The photoanode is transparent for red light that is absorbed by the porSi. Secondly, the changing pore diameter induces a larger porSi band gap near the illuminated side that becomes smaller near the dark side of the assembly. This gradient maximizes light absorption and drives photogenerated electrons to the dark side. There, carbon nanotube supported  $\text{CO}_2$  reduction catalysts are operated in the gas phase. Inside the porSi, proton conducting material is required that is permeable to gases to allow evacuation of the produced oxygen. At the dark side of the assembly, the pores are capped with gas-impermeable proton exchange material, in order to avoid product cross-over.

**Organic semiconductor solar membrane.** The knowledge gained from organic PV can be implemented in PEC cell research. By employing polymeric components based on *e.g.* PEDOT:PSS or P3HT:PCBM (poly(3-hexylthiophene): [6,6]-phenyl- $\text{C}_{60}$  butyric acid methyl ester), a mechanically flexible assembly is obtained. This could be important for portable PEC cells or to access a broader range of reactor designs. In Fig. 15, concentric assemblies are arranged in a multitubular set-up that could be submerged in aqueous solution. By feeding  $\text{CO}_2$  through the hollow tubes, the product is formed in a separate compartment and the  $\text{CO}_2$  stream is easily recycled.

A layered polymeric structure is ideally assembled *via* the LbL technique. LbL allows tuning the thickness of *e.g.* the proton conducting membrane for optimal gas separation yet minimal proton transport resistance. The anode and cathode thicknesses are chosen according to their light absorption and charge transport properties. An electrically conducting polymer could be the ideal backbone to immobilise  $\text{CO}_2$  reduction catalysts. By using a monomer with different functional groups, it is possible to precisely immobilise two or more catalytic complexes relative to each other (Fig. 16). Instead of a single catalytic centre, catalytic pockets can then be created in an 'artificial enzyme'. It has been shown that such simultaneous interaction with multiple catalytic species can be an effective strategy to activate the  $\text{CO}_2$  molecule.<sup>28</sup>

### 3. PEC characterisation methods

With PEC development still in its infancy, there is a lack of *in situ* characterisation methods. Conventional techniques

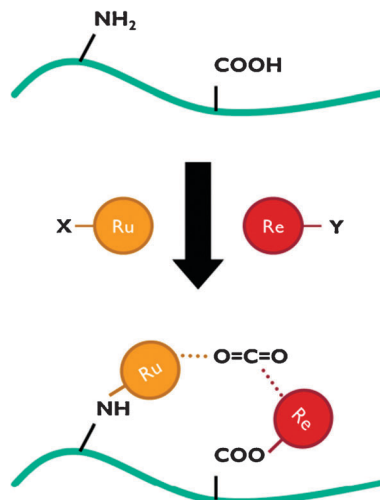


Fig. 16 Representation of an 'artificial enzyme' active site, obtained by selective immobilisation of Ru- and Re-polypyridyl complexes on specific functional groups.

often focus on individual cell components and consider only one of the half-reactions. For a deeper understanding, PEC cells should be monitored in real-time so both steady-state and transient effects can be investigated.

**Electrochemical techniques.** Monitoring electrical photo-currents and potentials in a wired PEC cell is straightforward. Typically, photoelectrodes are characterised by linear sweep and cyclic voltammetries in the dark or under illumination, in the presence of  $\text{CO}_2$  or under an inert atmosphere. From these data, information is obtained about *e.g.* charge recombination, reaction overpotential and catalytic rate constants.<sup>8,17</sup> These measurements are performed in a three-electrode cell containing a reference electrode, to avoid interference of events occurring at the counter electrode.<sup>17</sup> While this requirement is obvious for material scientists, it is not at all evident in PEC cell research. In conventional measurements, the overpotential of the working electrode is of interest. This overpotential may include contributions from *e.g.* reaction kinetics or diffusion limitations. The reference electrode serves as a reliable potential indicator which does not carry current, in order to avoid contributions from counter electrode overpotentials or ohmic losses through the electrolyte. When one is studying a complete cell, the use of a reference electrode interferes with the operation of the system. In these measurements, anode and cathode overpotentials and ohmic losses are in fact all of interest to the researcher. This means (photo)electrochemical measurements on a PEC cell should be carried out in a two-electrode set-up, *i.e.* without reference electrode. Obviously the influence of multiple components will complicate interpretation of the data. While electrochemical measurements on photoelectrodes in three-electrode measurements are common, two-electrode PEC cell voltammetry has not yet been investigated.

Such investigations have been carried out by some of the present authors for (two-electrode) PEC cell chronoamperometry (Fig. 17).<sup>24</sup> In PEC cell chronoamperometry, the current



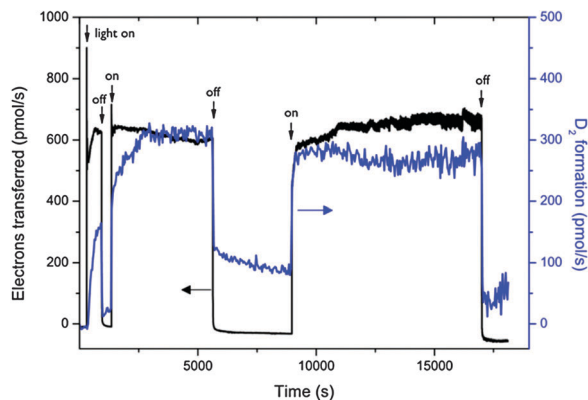


Fig. 17 Monitoring of photocurrent and  $D_2$  formation out of deuterated water in a fuel cell type solar membrane PEC cell with interrupted illumination. Reproduced from ref. 24 with permission from the PCCP owner societies.

is recorded as a function of time at a certain potential. Most measurements are performed at short circuit (0 V), but a bias may be applied. By shutting the illumination on and off, both transient effects and steady-state photocurrents can be investigated. The contributions of PEC cell components can be deconvoluted from the photocurrent profiles by modulating the reaction conditions or light source and comparing the response. For example, the addition of hydrogen peroxide as a hole scavenger will facilitate the anodic reaction and rule out any problems associated with the oxidation half-reaction. By introducing acid, the influence of proton transport on performance is largely eliminated. Importantly, overall PEC cell efficiencies are only realistic when obtained in a two-electrode set-up at short circuit, to avoid extracting additional power from the instrument. Chemical bias in two-compartment PEC cells can be avoided by keeping the pH of both sides identical. When using monochromatic light, the photon flux is known and the incident-photon-to-current-efficiency (IPCE or external quantum efficiency (EQE)) can be estimated:<sup>44</sup>

$$\text{IPCE} = \frac{\text{electron flux } [\text{mol s}^{-1}]}{\text{incident photon flux } [\text{mol s}^{-1}]} \quad (13)$$

The STF efficiency is obtained when using simulated solar light (typically Air Mass 1.5) at 1 sun intensity ( $100 \text{ mW cm}^{-2}$ ), or under concentrated solar illumination.<sup>44</sup> In addition, claims on STF efficiency should be based on quantification of formation rates of solar fuel and molecular oxygen. Provided the Faradaic Efficiency (FE) is known, STF efficiency can be calculated from the photocurrent:

$$\text{STF} = \frac{(E_{\text{H}_2\text{O}/\text{O}_2}^0 [\text{V}] - E_{\text{prod}}^0 [\text{V}]) \times I [\text{mA cm}^{-2}] \times \text{FE}}{P [\text{mW cm}^{-2}]} \quad (14)$$

of organic polymers.

Virtually any electrochemical characterisation method could be applicable to PEC cells and Electrochemical Impedance Spectroscopy (EIS) is especially useful. EIS has multiple uses for *in situ* PEC cell research. It makes use of a small modulated

voltage perturbation at different frequencies. The response of the PEC cell can be modelled as an equivalent circuit containing resistive (e.g. charge transport) and capacitive (e.g. semiconductor space charge layer) elements with different time constants. From the constructed model, information can be gathered about various processes including charge trapping, surface recombination and reaction kinetics.<sup>17</sup> Besides the properties of only one photoelectrode, overall PEC cell characteristics can be obtained in a two-electrode measurement. Information could be gathered about electrolyte and membrane transport resistance, and kinetic processes at the second electrode.<sup>45</sup>

**Product analysis.** Since the goal of  $\text{CO}_2$  reduction is to obtain chemical energy vectors, direct analysis of the chemicals produced in a PEC cell is the ultimate way to characterise PEC cell performance. Commonly, gas/ion chromatography and (chromatography coupled) mass spectrometry are used to quantify product formation. Since production rates are very low ( $\text{nmol s}^{-1}$  to  $\mu\text{mol s}^{-1}$  range), highly sensitive instruments are needed and control experiments have to show whether product formation is really associated with  $\text{CO}_2$  reduction, or due to impurities in the sample.<sup>9</sup> Batch reactors allow the build-up of product concentration, but the obtained data represent an integrated signal. In continuous flow reactors, more direct information is available. Since a mixture of products is often formed, a separation step is needed prior to detection which results in discrete data points.<sup>23,26</sup> If the selectivity of the reaction is sufficiently high, on-line mass spectrometry can be used (Fig. 17). The low production rates imply that high-purity media and well-sealed reaction chambers are to be used if one wants to get reliable data. In many cases, non-quantitative product formation is caused by practical problems rather than more fundamental issues.

Product detection is a direct way to measure PEC cell performance. If the electrical current is compared with production rates, product selectivity and Faradaic efficiency can be calculated, which are necessary to calculate STF efficiency.<sup>44</sup> Since PEC cell photocurrents are seldom constant, Faradaic efficiency is likely to be time-dependent as well.<sup>24</sup> Real-time product detection is required to determine whether features in photocurrent transients are associated with product formation or other, non-Faradaic processes. Moreover, in wireless PEC cells no current can be measured so product formation rate is the primary metric to characterise these systems.<sup>6,20</sup>

Mechanistic information can be obtained by modulating the illumination source and reagents. For example, catalytic activity can be investigated by cyclic voltammetry coupled to mass spectrometry, under  $\text{CO}_2$  or under Ar atmosphere and in the dark or under illumination.<sup>8</sup> Isotope tracer experiments (using  $^{13}\text{CO}_2$ ,  $\text{D}_2\text{O}$  or  $\text{H}_2^{18}\text{O}$ ) confirm the atom source and may exclude photocorrosion or reaction with impurities.<sup>6,24</sup>

**UV/Vis/NIR spectroscopy.** The prevalence of light in PEC cell research means spectroscopy is an often encountered characterisation technique. UV-Vis spectroscopy is frequently used for determining the semiconductor band gap.<sup>44</sup> Since reaction vessels are transparent, *in situ* spectroscopy can be implemented





rather easily. Infrared spectroscopy has been used for studying reaction intermediates on powdered samples.<sup>9</sup> For complete PEC cells, the combination of electrochemistry and spectroscopy (spectroelectrochemistry) is an interesting option. Smieja and Kubiak performed a spectroelectrochemical investigation of Re complexes for CO<sub>2</sub> reduction and observed dimerization of some compounds during electrolysis.<sup>28</sup> These experiments were performed on rather long time scales and provide information about the system at equilibrium at a certain bias potential. If transient effects and intermediate species are of interest, more advanced techniques can be used.

Transient Absorption Spectroscopy (TAS), also referred to as pump–probe spectroscopy or flash photolysis, relies on the instantaneous excitation of a system by an extremely short laser pulse (pump), followed by absorption spectroscopy using a second light source (probe). The probe can be a continuous illumination source or a time-delayed laser pulse, if events are to be studied on (sub-)nanosecond time scales. The decay of the excited species can be followed by the changes in absorption compared to the ground state of the system. By probing this absorption at various times after the pump pulse, a two-dimensional map is obtained as a function of wavelength and time. This time-resolved technique allows observation of species that are very short-lived. It has been used to look at charge carrier dynamics, reaction kinetics and catalytic cycles.<sup>12,13</sup> TAS gives direct information about hole dynamics, in contrast with electrochemical techniques which are based on the extraction of electrons from the photoelectrode. Tang *et al.* observed free and trapped hole populations in TiO<sub>2</sub>, and provided proof for a 4-hole water oxidation reaction.<sup>46</sup> Sakai and co-workers used TAS to investigate charge transfer within supramolecular complexes for CO<sub>2</sub> reduction.<sup>28</sup> While TAS on powder samples and homogeneous catalysts provides important insights, *in situ* measurements are needed that reliably reproduce the operating conditions of a PEC cell. Cowan *et al.* built a complete three-electrode PEC cell for TAS studies of nanocrystalline TiO<sub>2</sub> photoelectrodes.<sup>47</sup> They found that although slow water oxidation kinetics cause charge carrier recombination, the application of a positive bias leads to the removal of electrons and enhances the lifetime of photogenerated holes, which are then successfully used to oxidise water. In their system, separate electrodes were used on a transparent conducting glass substrate. Most monolithic PEC cells will be non-transparent, and will have to be used in a reflection set-up rather than in transmission mode. The combination of light excitation, electrochemical control and direct observation of charge carriers and intermediates makes *in situ* TAS an ideal technique for both fundamental studies and performance characterisation of PEC cells.

***In situ* nuclear magnetic resonance.** Nuclear Magnetic Resonance (NMR) is a spectroscopy based on nuclear spin transitions. As these transitions are influenced by the multiple interactions with the surroundings, NMR can probe many structural aspects of complex materials and even devices under working conditions. NMR has been used to determine the structure of organic CO<sub>2</sub> reduction catalysts<sup>8</sup> and to investigate the structural stability of water oxidation catalysts.<sup>2,20</sup> However, no *in situ* NMR studies on

working PEC cells have been performed yet. Scheler introduced NMR electrophoresis of polyelectrolyte membranes.<sup>48</sup> Grey and co-workers are using *in situ* NMR of lithium batteries during charge–discharge cycles.<sup>49</sup> Since NMR is a non-destructive technique to obtain atomic level information from complex systems, its *in situ* application is likely to provide fundamental insights into a working PEC cell. Possibilities include the observation of CO<sub>2</sub> conversions, proton transport in the catalyst structure or membrane, reagent transport and hydration of the membrane. *Ex situ* and *in situ* NMR studies can be usefully combined to gain atomic scale insights of the different parts of a PEC cell to further support and expand theoretical PEC cell models.

**PEC cell modelling.** Theoretical modelling is a relatively new approach to gain understanding of the complex processes that occur in the heart of a PEC cell. Simulation is an invaluable aid to direct research efforts and to gain fundamental understanding of the system. Modelling can assist in design, optimisation and scale-up of PEC cells and contribute to the identification of bottlenecks.

Models for hydrogen producing PEC cells have been developed, but so far models for the photoelectrochemical reduction of CO<sub>2</sub> are not available. Until recently, models for hydrogen producing PEC cells were mostly based on equivalent circuit models, introduced by Rocheleau and Miller.<sup>50</sup> These represent the PEC cell as an electrical circuit with an electrochemical load to account for ohmic losses and kinetic overpotentials. This theoretical approach has been widely used to model separate components of a PEC cell, *e.g.* to estimate photopotential as a function of semiconductor band gap in order to predict ideal light absorber materials or co-catalysts.<sup>45</sup> These models, however, are not detailed enough to grasp all features of a practical PEC cell. Multidimensional and multiphysics numerical modelling that includes charge and species conservation, accounts for electrochemical reaction kinetics and incorporates electron and ion transport phenomena are needed. An accurate model should allow identification of rate-limiting steps and indicate the improvements that can be expected with alternative materials, whether existing or hypothetical. Models incorporating spatial variables are useful to develop optimal reactor geometries.

Model-based device engineering is receiving more attention. Haussener *et al.* developed an adequate model based on mass and charge conservation, a Nernst–Planck description of species transport and Butler–Volmer equations for describing electrochemical kinetics.<sup>15</sup> This way they were able to simulate and compare different PEC cell geometries. They showed improved photocurrent densities by reduced ohmic losses when a transparent conducting oxide was added for current collection. Cross-over between compartments was calculated to be highest when the system was continuously flushed.

## Conclusions

The development of monolithic cells for solar fuel production is a lively research field, and solar hydrogen currently is the most popular topic. Progress is being made on the grand scientific



challenge of CO<sub>2</sub> reduction catalysts to produce solar carbon fuel, but the design of monolithic assemblies and PEC cells dedicated to CO<sub>2</sub> reduction is still in its infancy. PEC cell design is a complex matter and involves simultaneous managing of a multitude of processes related to illumination, charge separation, electrical conduction, molecular transport, catalytic chemistry, feed supply and reaction product recovery. This review has shown that these aspects should be an integral part of research on solar fuels. Promising concepts are the artificial leaf and the solar membrane. The sophistication of their design by implementing porosity and adopting concentric layered multimaterial assemblies will initiate a gradual transition from planar designs to truly three-dimensional structures. Atomic layer deposition and layer-by-layer assembly are enabling technologies to build future assemblies. The guidance by *in situ* time and space resolved spectroscopies and advanced modelling will path the way forward. The strategies discussed in this review are expected to lift the limited carbon-based solar-to-fuel efficiency closer to the levels of current solar hydrogen producing cells.

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## Notes and references

- 1 B. A. Pinaud, J. D. Benck, L. C. Seitz, A. J. Forman, Z. Chen, T. G. Deutsch, B. D. James, K. N. Baum, G. N. Baum, S. Ardo, H. Wang, E. L. Miller and T. F. Jaramillo, *Energy Environ. Sci.*, 2013, **6**, 1983–2002 and references therein.
- 2 K. S. Joya, Y. F. Joya, K. Ocaoglu and R. van de Krol, *Angew. Chem., Int. Ed.*, 2013, **52**, 10426–10437 and references therein.
- 3 A. Fujishima and K. Honda, *Nature*, 1972, **238**, 37–38.
- 4 J. Hemminger, R. Carr and G. Somorjai, *Chem. Phys. Lett.*, 1978, **57**, 100–104.
- 5 M. Halmann, *Nature*, 1978, **275**, 115–116.
- 6 T. Arai, S. Sato, T. Kajino and T. Morikawa, *Energy Environ. Sci.*, 2013, **6**, 1274–1282 and references therein.
- 7 B. Kumar, M. Llorente, J. Froehlich, T. Dang, A. Sathrum and C. P. Kubiak, *Annu. Rev. Phys. Chem.*, 2012, **63**, 541–569 and references therein.
- 8 E. E. Benson, C. P. Kubiak, A. J. Sathrum and J. M. Smieja, *Chem. Soc. Rev.*, 2009, **38**, 89–99 and references therein.
- 9 E. V. Kondratenko, G. Mul, J. Baltrusaitis, G. O. Larrazábal and J. Pérez-Ramírez, *Energy Environ. Sci.*, 2013, **6**, 3112–3135 and references therein.
- 10 M. Mikkelsen, M. Jørgensen and F. C. Krebs, *Energy Environ. Sci.*, 2010, **3**, 43–81.
- 11 M. Romero and A. Steinfeld, *Energy Environ. Sci.*, 2012, **5**, 9234–9245.
- 12 F. E. Osterloh, *Chem. Soc. Rev.*, 2013, **42**, 2294–2320 and references therein.
- 13 A. J. Cowan and J. R. Durrant, *Chem. Soc. Rev.*, 2012, **42**, 2281–2293 and references therein.
- 14 W. Fan, Q. Zhang and Y. Wang, *Phys. Chem. Chem. Phys.*, 2013, **15**, 2632–2649.
- 15 S. Haussener, C. Xiang, J. M. Spurgeon, S. Ardo, N. S. Lewis and A. Z. Weber, *Energy Environ. Sci.*, 2012, **5**, 9922–9935 and references therein.
- 16 J. M. Spurgeon, M. G. Walter, J. Zhou, P. a. Kohl and N. S. Lewis, *Energy Environ. Sci.*, 2011, **4**, 1772–1780.
- 17 R. Van de Krol and M. Grätzel, *Photoelectrochemical hydrogen production*, Springer, 2012 and references therein.
- 18 K. Maeda, M. Higashi, D. Lu, R. Abe and K. Domen, *J. Am. Chem. Soc.*, 2010, **132**, 5858–5868.
- 19 J. Newman, *J. Electrochem. Soc.*, 2013, **160**, 309–311.
- 20 S. Reece, J. Hamel, K. Sung, T. Jarvi, A. J. Esswein, J. J. H. Pijpers and D. G. Nocera, *Science*, 2011, **334**, 645–648 and references therein.
- 21 D. C. Grills and E. Fujita, *J. Phys. Chem. Lett.*, 2010, **1**, 2709–2718 and references therein.
- 22 P. Voyame and K. Toghill, *Inorg. Chem.*, 2013, **52**, 10949–10957.
- 23 C. Ampelli, G. Centi, R. Passalacqua and S. Perathoner, *Energy Environ. Sci.*, 2010, **3**, 292–301.
- 24 J. Rongé, D. Nijs, S. Kerkhofs, K. Masschaele and J. Martens, *Phys. Chem. Chem. Phys.*, 2013, **15**, 9315–9325.
- 25 M. G. Walter, E. L. Warren, J. R. McKone, S. W. Boettcher, Q. Mi, E. a. Santori and N. S. Lewis, *Chem. Rev.*, 2010, **110**, 6446–6473 and references therein.
- 26 S. Roy, O. Varghese, M. Paulose and C. Grimes, *ACS Nano*, 2010, **4**, 1259–1278.
- 27 Y. Oh and X. Hu, *Chem. Soc. Rev.*, 2013, **42**, 2253–2261.
- 28 M. Schulz, M. Karnahl, M. Schwalbe and J. G. Vos, *Coord. Chem. Rev.*, 2012, **256**, 1682–1705 and references therein.
- 29 R. E. Rocheleau, E. L. Miller and A. Misra, *Energy Fuels*, 1998, **12**, 3–10.
- 30 S. Ichikawa and R. Doi, *Catal. Today*, 1996, **27**, 271–277.
- 31 P. Zhai, S. Haussener and J. Ager, *Energy Environ. Sci.*, 2013, **6**, 2380–2389.
- 32 Y. Surendranath, D. K. Bediako and D. G. Nocera, *Proc. Natl. Acad. Sci. U. S. A.*, 2012, **109**, 15617–15621.
- 33 P. Menna, G. Di Francia and V. La Ferrara, *Sol. Energy Mater. Sol. Cells*, 1995, **37**, 13–24.
- 34 L. Boarino, E. Enrico, N. De Leo, F. Celegato, P. Tiberto, K. Sparnacci and M. Laus, *Phys. Status Solidi A*, 2011, **208**, 1403–1406.
- 35 M. Ernst and R. Brendel, *Phys. Status Solidi*, 2012, **6**, 187–189.
- 36 S. Moghaddam and E. Pengwang, *Nat. Nanotechnol.*, 2010, **5**, 230–236.
- 37 E. Garrone, F. Geobaldo, P. Rivolo, G. Amato, L. Boarino, M. Chiesa, E. Giamello, R. Gobetto, P. Ugliengo and A. Viale, *Adv. Mater.*, 2005, **17**, 528–531.



- 38 E. Seyrek and G. Decher, in *Polymer Science: A Comprehensive Reference*, ed. K. Matyjaszewski and M. Möller, Elsevier, 2012, pp. 159–185 and references therein.
- 39 A. Izquierdo, S. S. Ono, J.-C. Voegel, P. Schaaf and G. Decher, *Langmuir*, 2005, **21**, 7558–7567.
- 40 J. Hiller, J. D. Mendelsohn and M. F. Rubner, *Nat. Mater.*, 2002, **1**, 59–63.
- 41 C. Detavernier, J. Dendooven, S. P. Sree, K. F. Ludwig and J. A. Martens, *Chem. Soc. Rev.*, 2011, **40**, 5242–5253.
- 42 Q. Peng, J. S. Lewis, P. G. Hoertz, J. T. Glass and G. N. Parsons, *J. Vac. Sci. Technol., A*, 2012, **30**, 010803.
- 43 P. Poodt, D. C. Cameron, E. Dickey, S. M. George, V. Kuznetsov, G. N. Parsons, F. Roozeboom, G. Sundaram and A. Vermeer, *J. Vac. Sci. Technol., A*, 2012, **30**, 010802.
- 44 Z. Chen, T. F. Jaramillo, T. G. Deutsch, A. Kleiman-Shwarscstein, A. J. Forman, N. Gaillard, R. Garland, K. Takanabe, C. Heske, M. Sunkara, E. W. McFarland, K. Domen, E. L. Miller, J. A. Turner and H. N. Dinh, *J. Mater. Res.*, 2011, **25**, 3–16.
- 45 T. Lopes, L. Andrade, H. A. Ribeiro and A. Mendes, *Int. J. Hydrogen Energy*, 2010, **35**, 11601–11608.
- 46 J. Tang, J. R. Durrant and D. R. Klug, *J. Am. Chem. Soc.*, 2008, **130**, 13885–13891.
- 47 A. J. Cowan, J. Tang, W. Leng, J. R. Durrant and D. R. Klug, *J. Phys. Chem. C*, 2010, **114**, 4208–4214.
- 48 U. Scheler, *Curr. Opin. Colloid Interface Sci.*, 2009, **14**, 212–215.
- 49 R. Bhattacharyya, B. Key, H. Chen, A. S. Best, A. F. Hollenkamp and C. P. Grey, *Nat. Mater.*, 2010, **9**, 504–510.
- 50 R. E. Rocheleau and E. L. Miller, *Int. J. Hydrogen Energy*, 1997, **22**, 771–782.

