

# Photophoretic separation of single-walled carbon nanotubes: a novel approach to selective chiral sorting†

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For over two decades single-walled carbon nanotubes (SWCNTs) have been used in a broad range of electronic and optical applications, however the selective chiral sorting of SWCNTs with guaranteed optoelectronics characteristics is imperative to the industrial realization of such applications. In this paper we provide the results of modeling an optical sorting method that utilizes the inherent opto-electronic properties of the SWCNTs, thus guaranteeing the properties of the extracted populations. Utilizing the resonant transfer of photonic momentum, we simulate chiral sorting of two chiral populations in an aqueous environment based on the frequency dependent optical absorption properties of the nanotubes. We show that photonic sorting is not only feasible, but may be up to faster than density gradient centrifugation techniques. Our simulations investigate the effects of laser power, temperature and orientation. We find that 96% purity can be achieved in less than 12 minutes by operating at  $9 \times 10^7 \text{ W m}^{-2}$  (20 mW in a 20  $\mu\text{m}$  chamber) at elevated temperatures.

## Introduction

Since they were first observed in 1991,<sup>1</sup> many different applications have been proposed for carbon nanotubes including energy conversion,<sup>2–4</sup> chemical and optical sensors,<sup>5–7</sup> hydrogen storage,<sup>8–11</sup> high-speed electronics<sup>12–16</sup> and computing,<sup>17</sup> nanometer wires and high strength composites.<sup>18</sup> In recent years some applications have been commercially realized but these are mainly composite materials which have little dependence upon diameter or chirality. Although advanced optoelectronic applications have been demonstrated<sup>2</sup> the lack of bulk chiral selectivity with guaranteed optoelectronic properties means these are, generally, still confined to the laboratory. The electronic and optical properties of single walled carbon nanotubes (SWCNTs) intrinsically depend upon their chirality due the graphene wrapping.<sup>19</sup> The chirality of a carbon nanotube is described by two integer values ( $m$  and  $n$ ), its chiral vector (the dashed line in Fig. 1) that describes how it is rolled from a graphene sheet.

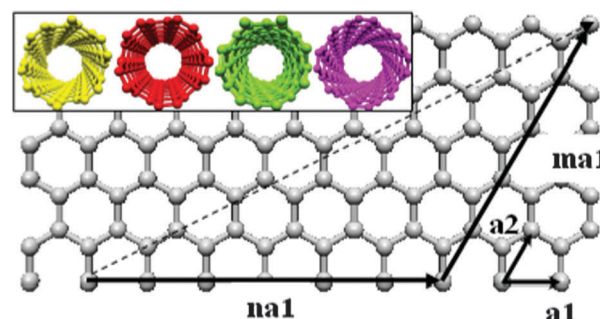


Fig. 1 Graphene sheet showing how the chiral vector (dashed arrow) of a (6,5) carbon nanotube is described by two integer values  $m$  and  $n$ . Inset: end-on-view of various chirality carbon nanotubes; from left to right, (6,4), (5,5), (7,1), (8,4).

For many advanced applications mono-chiral dispersions are essential<sup>20</sup> and a range of post-synthetic sorting techniques have been developed to achieve this. These techniques include density gradient ultra-centrifugation,<sup>21–30</sup> ion exchange chromatography,<sup>31</sup> gel chromatography<sup>32</sup> and conjugated polymer extraction.<sup>33</sup> However, density gradient ultracentrifugation is the most commonly used. The density gradient ultracentrifugation of nanotubes was first demonstrated by Arnold *et al.*<sup>21</sup> in 2006 who showed diameter control the of  $\sim 0.02 \text{ nm}$ . Significant effort was subsequently invested in developing the technique. Fagan *et al.*<sup>25</sup> demonstrated that density gradient ultracentrifugation could be used for SWCNT length fractionation whilst numerous other groups have demonstrated that the technique

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can be used two separate metallic and semiconductor SWCNT.<sup>21,23,24</sup> More recently chiral purity exceeding 97% have now been obtained *via* DGU<sup>22</sup> and some groups have even demonstrated enantiomeric selectivity.<sup>26,28</sup> However processing times limit the scalability of density gradient ultracentrifugation. For example the procedure originally used by Arnold *et al.*<sup>21</sup> in 2006 required 12 hours of DGU processing time whereas the more recent enantiomeric selectivity demonstrated by Ghosh *et al.*<sup>28</sup> required 18 hours of processing.

Chiral recognition and chromatographic techniques have also been vigorously investigated. In 2007 Nish *et al.*<sup>33</sup> demonstrated that specific aromatic polymers would selectively disperse particular SWCNT species. Whilst this is a relatively simple method the resulting dispersions still only demonstrated a chiral purity of ~60%. In 2009 Tu *et al.*<sup>31</sup> used custom fabricated DNA sequences that preferentially encapsulate specific chiralities depending upon the DNA sequence. The encapsulation allowed for the selective chemical modification and separation using ion-exchange chromatography. However, such an approach may not be commercially viable due to the prohibiting cost of DNA strand synthesis and (if required) the subsequent difficulties with DNA removal. More recently, a simple gel chromatographic method of sorting was demonstrated by Liu *et al.*<sup>32</sup> The procedure is able to refine SWCNTs based upon their interaction with the allyl dextran-based size-exclusion gel. The technique is clearly very powerful and yet careful study of the photoluminescence result still demonstrates the presence of other chiralities within the refined samples.

Although both of the aforementioned techniques are very powerful tools for producing mono-chiral dispersions they cannot guarantee the optoelectronic characteristics of the purified SWCNTs (merely their chirality) due to the indirect approaches they adopt for chiral recognition. To sort high quality, defect free SWCNTs a direct method based upon optoelectronic interactions would be desirable. The simplest method used to characterize poly-dispersed carbon nanotube suspensions is UV-Vis-NIR absorption spectroscopy. This optical technique gives an indication to the chiralities of SWCNTs present in a sample, based on the exact frequency or wavelength of their absorption peaks. In the following work, we demonstrate using a computer simulation how it is possible to use absorption as a tool for separation rather than just for characterization.

The potential to use the absorption of photons for separation is due to photons possessing an amount of momentum based upon their frequency, and hence their energy. Therefore, if one chirality of SWCNTs absorbs more photons at a given energy than a different chirality, the preferentially absorbing sample will experience a greater photon pressure. Examination of the density of states (DOS) of SWCNTs shows sharp spikes, which are Van Hove singularities<sup>34</sup> and occur at energies dependent upon the chirality of the carbon nanotubes. The peaks in the absorption spectra are due to optically excited transitions preferentially occurring at energies coinciding with transitions between Van Hove singularities. Hence, by tuning a photonic source, for example a laser, to the transition energy between corresponding Van Hove singularities, it should be possible to preferentially impart photonic momentum upon a single chirality of SWCNTs.

A fundamental question to ask of such a separation strategy would be whether sufficient photonic momentum can be imparted on the SWCNT populations to overcome Brownian based diffusive re-mixing without optically saturating the transitions. The work presented here demonstrates that this is possible and has been tested by simulations run under a number of possible conditions to elucidate the required separation times of two populations of SWCNTs. We have chosen to use a simulations rather than an analytical approach to allow for the SWCNTs reduced freedom of rotation due to interaction with the channel sidewalls.

## Theory

Brownian movement (*i.e.* diffusion) of SWCNTs has both rotational and translational components and here we base our simulation upon previously published work.<sup>35</sup> Brownian translation was considered as a Gaussian distribution of random steps in the *x*, *y* and *z* directions. Such a distribution would have a mean of zero and a standard deviation of  $\sigma$ .  $\sigma$  can be found using eqn (1).

$$\sigma^2 = \langle (B_{x,y,z})(B_{x,y,z}) \rangle = 2D_{x,y,z}\Delta t \quad (1)$$

$B_{x,y,z}$  are the mean absolute step sizes along the respective *x*, *y* and *z* directions while  $D_{x,y,z}$  are the associated diffusivity constants along these directions.  $\Delta t$  is the time step of the simulation. The diffusivity constants were governed by the ratio between thermal energy and the hydrodynamic friction. This can be seen in eqn (2):

$$D_{x,y,z} = k_B T \zeta_{x,y,z}^{-1} \quad (2)$$

$k_B$  is Boltzmann constant,  $T$  is temperature in Kelvin and  $\zeta_{x,y,z}$  are the hydrodynamic friction coefficients in the *x*, *y* and *z* directions. The hydrodynamic friction coefficients can be found from the eigenvalues of the hydrodynamic friction tensor given in eqn (3):

$$\underline{\underline{\zeta}} = \underline{\underline{\zeta}}^{\parallel} \underline{\underline{U}} + \underline{\underline{\zeta}}^{\perp} \left( \underline{\underline{I}} - \underline{\underline{U}} \right) \quad (3)$$

$\underline{\underline{U}}$  represents a unit vector pointing along each SWCNT axis and  $\underline{\underline{I}}$  the identity matrix.  $\underline{\underline{\zeta}}^{\parallel}$  and  $\underline{\underline{\zeta}}^{\perp}$  are the hydrodynamic drag coefficients parallel and perpendicular the SWCNT axis given in eqn (4) and (5).

$$\underline{\underline{\zeta}}^{\parallel} = \frac{2\pi\eta l_{nt}}{\ln\left(\frac{l_{nt}}{2(r_{nt} + \alpha)}\right)} \quad (4)$$

$$\underline{\underline{\zeta}}^{\perp} = 2\underline{\underline{\zeta}}^{\parallel} \quad (5)$$

$l_{nt}$  and  $r_{nt}$  are the SWCNT length and radius respectively.  $\eta$  is the hydrodynamic viscosity.  $\alpha$  is an additional radius used to allow for a hydration boundary such that the total radius  $r_{nt} + \alpha$  represents the hydrodynamic radius.

Brownian rotation was modeled by finding the standard deviation in two orthogonal planes normal to the SWCNT axis. The rotational standard deviation is given by:

$$\sigma^2 = \langle (B_{UV,UW})(B_{UV,UW}) \rangle = 2D_R\Delta t \quad (6)$$

In the eqn (6),  $B_{UV}$  and  $B_{UW}$  are the mean absolute rotational steps in the two planes normal to the SWCNTs' axis.



Eqn (7) shows how  $D_R$  is the rotational diffusivity and which can be found from the ratio of the thermal energy to the rotational hydrodynamic frictional coefficient  $\xi^R$ , where  $\xi^R$  is obtained from eqn (8):

$$D_R = k_B T / \xi^R \quad (7)$$

$$\xi^R = \frac{\pi \eta l_{nt}}{\ln\left(\frac{l_{nt}}{2(r_{nt} + \alpha)}\right)} \times \left( \frac{1 + 0.64 \ln\left(\frac{l_{nt}}{2(r_{nt} + \alpha)}\right)}{1 + 1.5 \ln\left(\frac{l_{nt}}{2(r_{nt} + \alpha)}\right)} + 1.659 \left( \ln\left(\frac{l_{nt}}{2(r_{nt} + \alpha)}\right) \right)^2 \right) \quad (8)$$

Many studies have been conducted both experimentally<sup>36–38</sup> and theoretically<sup>39,40</sup> into the behaviour of asymmetric objects in optical traps. However, optically traps typically exhibit strong degrees of polarisation and intensity gradients producing forces that both confine and orient objects. Conversely, in our case such forces could be a hindrance. Therefore, we assume an unpolarised, uniform optical field (*i.e.* no optical gradients) of relatively low intensity. The photonic translation caused by the momentum transfer from absorbed photons is assumed to be along the direction of the Poynting vector (*i.e.* along the direction of propagation) and for convenience this is chosen to be the  $z$ -direction with no components in  $x$  or  $y$ . Absorption in SWCNTs arises due to many phenomena, including plasmon absorption, lattice scattering and photoelectric absorption. The first two effects are non-specific and non-saturable, creating a universal background absorption irrespective of chirality. However, the degree of photoelectric absorption at a given energy depends upon the SWCNT chirality. Further to this, it has been shown that the photoelectric absorption of SWCNTs is a saturable process with a continuous wave saturation intensity of  $\sim 5 \times 10^7 \text{ W m}^{-2}$ .<sup>41</sup>

As such, the energy absorbed ( $\psi$ ) by a partially saturable absorbing SWCNT in a time period  $\Delta t$  is given by:

$$\psi = \frac{I(x, y, z) \sigma(\theta, \phi)}{\tau} \left( 1 + \frac{\tau - 1}{1 + \frac{I(x, y, z)}{I_0}} \right) \quad (9)$$

where  $\tau$  is the ratio<sup>42,43</sup> of the total absorption (non-specific background plus photoelectric) to the background absorption.  $I(x, y, z)$  is the local energy flux at the position of the SWCNT.  $\sigma(\theta, \phi)$  is the SWCNT geometric projection in  $z$  which depends upon the SWCNT azimuthal ( $\theta$ ) and polar ( $\phi$ ) orientation and  $I_0$  is the saturation intensity. In this case when  $I(x, y, z) \ll I_0$  eqn (9) tends towards  $\psi = I(x, y, z) \sigma(\theta, \phi)$  which represents unsaturated absorption  $\sim \text{flux} \times \text{area}$ . When  $I(x, y, z) \gg I_0$  only non-specific absorption occurs and eqn (9) tends towards  $\psi = I(x, y, z) \sigma(\theta, \phi) / \tau$ . Here the total energy absorbed is decreased by  $\tau$ .

Once the total absorbed energy has been deduced the associated momentum change in the  $z$ -direction ( $P_z$ ) can be found from eqn (10) where  $c$  is the speed of light.

$$P_z = \frac{\psi}{c} \quad (10)$$

This momentum acts against the hydrodynamic frictional drag coefficient in the  $z$ -direction obtained in the Brownian translation calculations. The resulting photonic step is given by eqn (11):

$$B_z = P_z \cdot \xi_z \quad (11)$$

## Methods

All simulations were run on the Bluecrystal HPC facility at the University of Bristol. Parallel implementation was realized by utilizing the Python MPI4Py module. A typical simulation utilized 3 nodes with 8 cores per node such that there were 24 cores in total. The general procedure for these simulations can be found in the ESI† in Fig. S1. The SWCNTs are distributed evenly across all cores and are allowed to evolve for 54 000 cycles (54 seconds) before being re-gathered and their positions stored. This improved efficiency of the simulation by minimizing the amount of MPI communication required. Consequently the efficiency scaling is essentially linear; doubling the number of cores halves the processing time. 54 seconds is chosen between storage steps producing 200 data points during a 180 minute simulation.

The general experimental parameters can also be found in the ESI† in Table S1. Briefly, a 1 ms time step is chosen because during this time SWCNTs typically rotate less than 10 degrees allowing for a stable simulation. A simulated 'Run Time' of 180 minutes is needed to observe chiral separation. SWCNT lengths and diameters were based on SG65i SWCNTs which are commercially available from both Sigma-Aldrich and Southwest Nanotechnologies. The additional radial value is chosen so that hydrodynamic radius (which may be significantly larger than the SWCNT radius due to the presence of dispersion agents) is equal that of Sodium-dodecylbenzenesulfonate (SDBS) encapsulated SWCNTs. SDBS which has been shown to be one of the best surfactants for SWCNT dispersion.<sup>44</sup> Surfactant encapsulated SWCNTs have a larger hydrodynamic radius than SWCNTs dispersed *via* DNA or polymer encapsulation. Therefore, we allow for the 'worst case scenario'. The absorption-ratio was chosen by studying absorption spectra.<sup>42,43</sup> Low SWCNT concentrations were assumed and therefore SWCNT-SWCNT hydrodynamic coupling effects were considered negligible and ignored. We also model our samples as individual SWCNTs; as such they are not bundled. Assuming individual SWCNTs present at low concentrations means that any inter-SWCNT interactions, such as Exciton Energy Transfer (EET) or Fluorescence resonance energy transfer (FRET) can be disregarded. The potential experimental implication of EET or FRET within small bundles is considered in the discussion below.

We investigated a photophoretic<sup>45,46</sup> system due to the simple setup geometry and success of other phoretic approaches in



chiral processing.<sup>47</sup> Before conducting these investigations we verified the validity of our program by running Brownian evolution simulations of randomly orientated SWCNTs that can translate & rotate, and of aligned SWCNTs that can only undergo translation. We then analysed the results to obtain diffusion coefficients of aligned and randomly orientated SWCNTs and compared our findings with published results.

The simulated photophoretic system used a single resonantly tuned laser lightly focused into a  $20 \times 20 \times 1500 \mu\text{m}$  channel. To assess channel sidewall interactions limiting the SWCNTs' freedom of rotation and subsequent absorption cross section we initially simulated the Brownian motion of 10 000 SWCNTs over a 250 second period. We then recorded their absorption cross section as a function of distance from the channel sidewall.

To assess the effectiveness of photophoretic chiral sorting a mixed population of SWCNTs (50% on-resonance:50% off-resonance) was then introduced at one end of the a similar channel so that the photonic pressure drove the suspended SWCNTs along the channel.

When considering the following simulations, it is more useful to consider photonic intensity rather than absolute laser power. As such, laser powers are measured in terms of their relative photonic intensity (RPI); the ratio of photonic intensity in the sample chamber to a previously measured saturation intensity<sup>41</sup> of  $5 \times 107 \text{ W m}^{-2}$ . Therefore, an RPI of 1.0 equates to a photonic intensity which is the same as the saturation intensity.

Accordingly an RPI of 2.0 represent a photonic intensity which is double the saturation intensity. In this way an RPI of 1.0 corresponds to a laser power of 20 mW uniformly distributed across the modelled channel. Initially we verified that photophoretic separation of SWCNTs is viable and compared the separation times of aligned and randomly orientated dispersions. Alignment perpendicular to the z-axis should increase the SWCNT absorption cross-section and therefore may improve the separation efficiency. These simulations were conducted with an RPI of 1.0 (*i.e.* the saturation intensity) and at room temperature ( $20^\circ\text{C}$ ).

Subsequently, we conducted a power sweep at a constant temperature of  $20^\circ\text{C}$ , and a temperature sweep at a constant RPI of 1.0. For the power sweep the RPI was increased in steps of 0.25 between 0.25 and 2.5, whilst for the temperature sweep the solution temperature was increased from  $0^\circ\text{C}$  to  $100^\circ\text{C}$  in  $10^\circ\text{C}$  steps. We also allowed for the decrease in the hydrodynamic viscosity with increasing temperature *via* an empirical relationship. This relationship and a corresponding table of values can be found in the ESI† in eqn (S1) and Table S2.

At elevated temperatures surfactant encapsulation may not be stable method of SWCNT dispersal. However other dispersal methods are valid at elevated temperatures. For example it has been shown<sup>48</sup> that 12-base single-stranded DNA homopolymers of Guanine can disperse SWCNTs at temperature exceeding  $90^\circ\text{C}$ .

Finally, we conducted a series of simulations to ascertain the optimum separation conditions in terms of the RPI and temperature. This variable space assessment was conducted by setting the RPI and temperature to all possible combinations.

The respective ranges and increments were the same as for the RPI and temperature sweeps discussed above. Thus the full assessment consists of 110 simulations.

## Results

Our approach was validated by simulating diffusion caused by the translation of SWCNTs due to their Brownian motion. Fig. 2a and b shows the output of the simulation of randomly orientated SWCNTs before and after 5 seconds of free diffusion, with their time dependent mean square displacement (MSD) along the x, y and z axes plotted in Fig. 2c.

Fig. 2d and e show aligned SWCNTs before and after 5 seconds of free diffusion with their time dependent MSD shown in Fig. 2f. The related linear fits for each axis x, y and z gave diffusion constants of  $0.732$ ,  $0.693$  and  $0.657 \mu\text{m}^2 \text{s}^{-1}$  for randomly orientated SWCNTs. For the SWCNTs that are aligned parallel to the z-axis and not allowed to rotate, the diffusion constants are  $0.599$ ,  $0.614$  and  $1.21 \mu\text{m}^2 \text{s}^{-1}$  along the respective x, y and z axes. These results are in good agreement with the literature.<sup>22</sup> Fig. 2g shows the how the effect of the channel sidewall limits the SWCNT's freedom of rotation and the mean

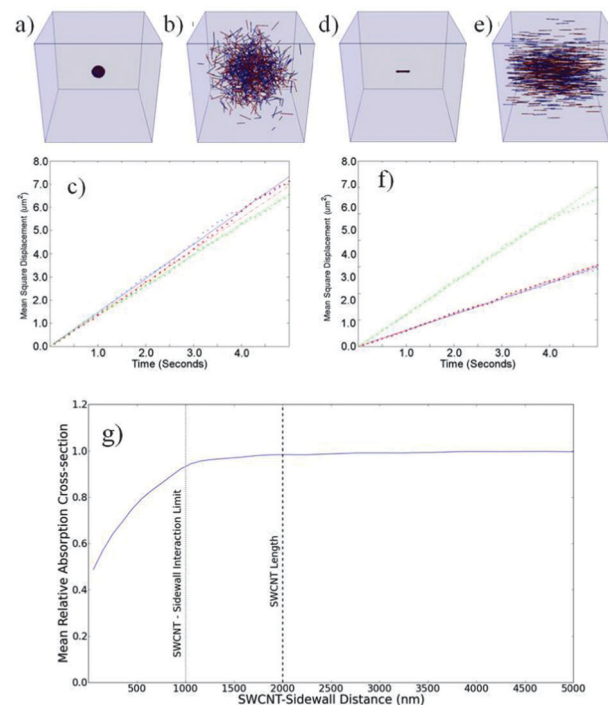
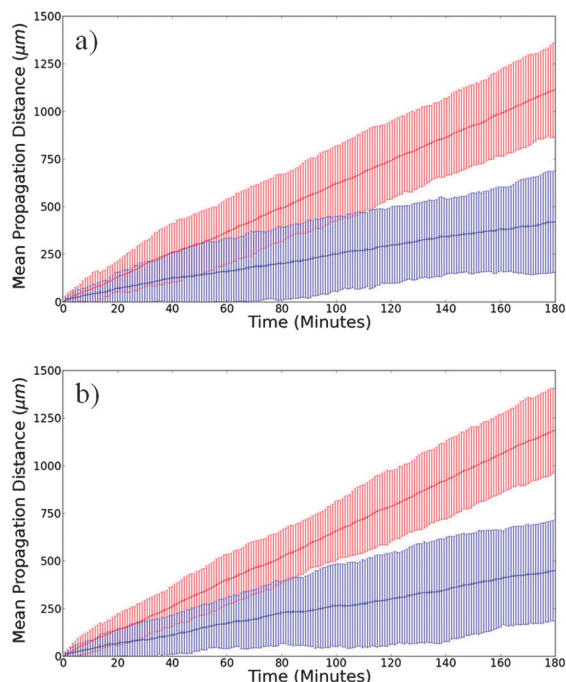


Fig. 2 Verification of translational Brownian simulations (a–f) and sidewall effects (g). The box dimensions (a–d) are  $16 \mu\text{m} \times 16 \mu\text{m} \times 16 \mu\text{m}$ . Randomly orientated SWCNTs (a) before and (b) after 5 seconds of evolution. (c) Mean square displacement (MSD) as a function of time for randomly orientated, rotationally free SWCNTs in the x (blue), y (red) and z (green) directions, (d–e) aligned SWCNTs (d) before and (e) after 5 seconds of evolution. (f) MSD as a function of time for each axis when non-rotating SWCNTs are aligned parallel to the z-axis, with x (blue), y (red) and z (green). The linear fits shown in (c) and (f) are used to obtain 1-dimensional diffusion constants. (g) Mean SWCNT absorption cross-section as a function of SWCNT (centre point)-sidewall separation.







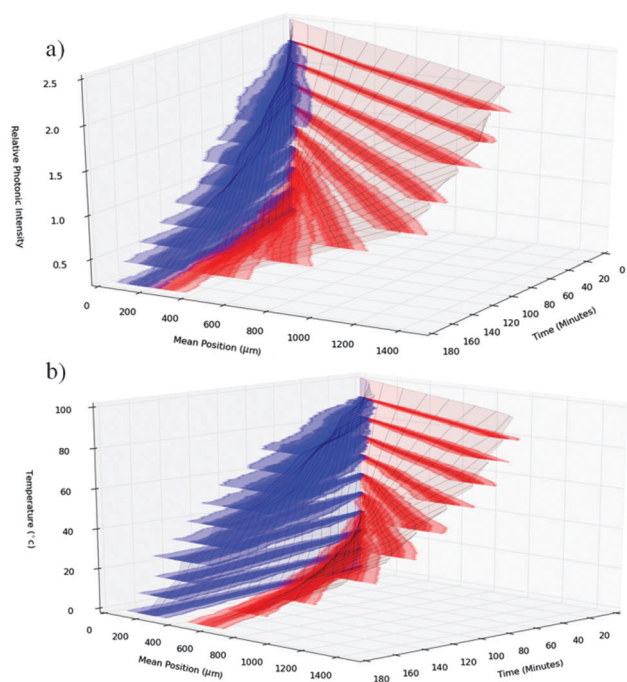
**Fig. 3** Mean position as a function of time of on-resonance (red) and off-resonance (blue) SWCNT populations undergoing photophoretic separation. (a) Distribution of randomly orientated SWCNT dispersions; (b) distribution of aligned SWCNT dispersions. The error bars are  $2\sigma$  and thus encompass 96% of each population. The  $2\sigma$ -separation times are 84 minutes and 109 minutes respectively.

hence absorbing cross section. On the average the sidewall effects reduced the absorbing cross section by over a range of  $\sim 1.2 \mu\text{m}$ .

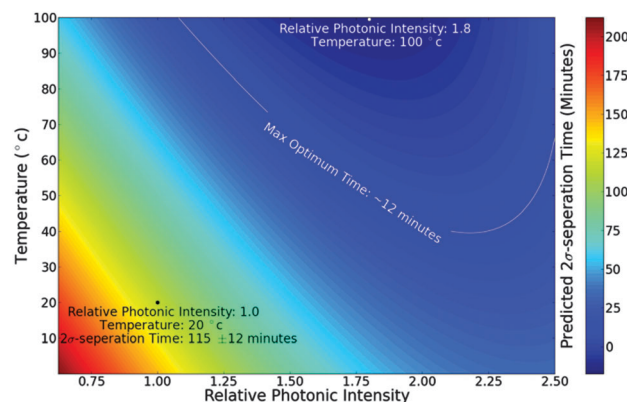
Fig. 3a shows photophoretic chiral separations of randomly orientated SWCNTs and Fig. 3b shows analogous data for aligned SWCNTs. Randomly orientated SWCNTs achieved  $2\sigma$ -separation after  $\sim 84$  minutes of processing at a distance of  $\sim 400 \mu\text{m}$  from their starting location. Aligned SWCNT achieved  $2\sigma$ -separation after  $\sim 109$  minutes at a distance of  $480 \mu\text{m}$ .

Fig. 4a shows stacked distribution plots of randomly oriented SWCNTs as a function of increasing RPI; the stacked distribution plots of randomly oriented SWCNTs as a function of temperature are shown in Fig. 4b. The increasing angular separation shown between the off-resonance SWCNTs (blue) and the on-resonance SWCNT (red) regions indicates an increased rate of separation both with increasing RPI or increasing temperature.

Finally, Fig. 5 shows the variable space simulations surface plot of RPI against temperature of predicted  $2\sigma$ -separation times for randomly aligned SWCNTs obtained *via* a 'best-fit' 2nd order polynomial surface. This found a minimum  $2\sigma$ -separation time of  $\sim 9$  minutes at a distance of  $280 \mu\text{m}$  with an RPI of 1.75 and a solution temperature of  $100^\circ\text{C}$ . As a general trend the  $2\sigma$ -separation times decrease with increasing temperature and RPI. At low temperatures and RPI some simulations never achieve  $2\sigma$ -separation within the 180 minutes simulated period.



**Fig. 4** Stacked distribution plots showing: (a) increasing separation rates (at a constant solution temperature of  $20^\circ\text{C}$ ) with increasing relative photonic intensity (RPI) and (b) increasing separation rates (at constant RPI of 1.0) with increasing temperature. The RPI is a measure of the photonic intensity relative to the saturation intensity. Thus an RPI of one represents the saturation intensity.



**Fig. 5** Relative photonic intensity (RPI) – temperature surface plot of predicted  $2\sigma$ -separation times for randomly aligned SWCNTs obtained *via* a 'best-fit' 2nd order polynomial surface. The error of  $\pm 12$  minutes was found from the standard deviation of simulated and predicted values. The black dot corresponds to the result shown in Fig. 3b. The white dot as at the surface minima indicates the conditions for optimum separation. The white contour line represents the 12 minute error bound of the optimum conditions and is the suggested experimental region.

## Discussion

The validity of our simulation was checked by: (i) estimating the diffusion constant of randomly aligned SWCNTs over a 5 second period compared to published values<sup>49</sup> and (ii) checking that aligned SWCNT diffuse twice as readily along their axis



of alignment as they do perpendicular to it.<sup>20</sup> As the  $x$ ,  $y$  and  $z$  axes can be considered equivalent, one would expect the 1-dimensional diffusion constants should be the same along all axes for the randomly orientated SWCNTs and similarly along the  $x$  and  $y$  directions for the aligned SWCNTs. The randomly orientated SWCNTs diffusion constant varies from  $0.657$  to  $0.732 \mu\text{m}^2 \text{s}^{-1}$  with a mean of  $0.694 \mu\text{m}^2 \text{s}^{-1}$ . For the aligned SWCNTs the diffusion constants are  $0.599$  and  $0.614$  with a mean of  $0.607 \mu\text{m}^2 \text{s}^{-1}$ . We therefore estimate that our uncertainty of the 1-dimensional diffusion constant is less than  $0.05 \mu\text{m}^2 \text{s}^{-1}$ . The 3-dimensional diffusion constant is then calculated for randomly aligned SWCNTs from the Euclidian norm of the 1-dimensional diffusion constants. This is found to be  $1.21 \pm 0.08 \mu\text{m}^2 \text{s}^{-1}$ , which falls within the range of previously measured values.<sup>49</sup> For the SWCNTs aligned to the  $z$ -axis, the  $z$ -axis diffusion constant ( $1.2 \mu\text{m}^2 \text{s}^{-1}$ ) is twice that of the  $x$ -axis ( $0.599 \mu\text{m}^2 \text{s}^{-1}$ ) and  $y$ -axis ( $0.614 \mu\text{m}^2 \text{s}^{-1}$ ) diffusion constants. This agrees with theoretical results,<sup>50</sup> further reassuring us as to the validity of our simulation. Our investigations into the effect of the sidewalls limiting the absorption cross-section found that the mean cross section fell gradually from  $\sim 95\%$  (at  $1200 \text{ nm}$  separation) to  $\sim 50\%$  (at  $100 \text{ nm}$  separation) of the bulk value. This is interesting as the separation at which the SWCNTs no longer interact with the sidewall is  $1000 \text{ nm}$  (*i.e.* the  $1/2$  length). However, the absorption cross-section doesn't recover to within  $5\%$  of the bulk value for a further  $200 \text{ nm}$ . Suggesting a degree of orientation memory. Considering how the sidewall effect may alter the photophoretic behaviour we calculate that over the  $20 \mu\text{m} \times 20 \mu\text{m}$  channel the mean SWCNT absorption cross-section is only reduced by  $6\%$  and compared to no sidewall interaction and should therefore have a limited effect on the sorting efficiency.

Initial testing on the photophoretic separation gave rise to the results shown in Fig. 3a and b, which confirm that SWCNT chiral separation can be achieved in less than 2 hours for both aligned and randomly distributed populations. Aligned SWCNT displayed shorter  $2\sigma$  separation times due to their increased cross-section in the  $xy$  plane and were found to achieve  $2\sigma$  separation in approximately three quarters of the time that it took for randomly orientated SWCNTs to achieve the same degree of separation. Whilst this effect was expected, the enhancement is significantly less than had been originally envisaged and suggests that alignment procedures would only offer marginal gains to separation. Consequently, potential future separation devices do not need to include alignment mechanisms and therefore allow for simpler and more elegant approaches to be considered.

Fig. 4 shows increasing rates of separation when both the RPI (Fig. 4a) and the temperature (Fig. 4b) are increased. The separation rates increase with RPI because the photonic 'pushing' effect becomes stronger at higher RPIs. However, at RPIs above  $1.0$  the absorption band will saturate. Therefore it should be more difficult to discriminate between the on-resonance and off-resonance SWCNTs. The reduced discrimination will result in slower separation rates. A possible explanation for not observing slower separation rates at higher RPIs, is that the

effect of saturation limiting the separation may only become apparent at RPIs greater  $2.5$  which was the limit of what was simulated here.

The other interesting observation is that separation rates increase with temperature. As the diffusivity is proportional to temperature (see eqn (2)), one would expect the diffusion driven remixing of the SWCNTs to be greater at higher temperatures and hence slow the overall rate of separation. However, this is not the case. One also has to consider the change in viscosity with temperature to understand this observation. Between  $0^\circ\text{C}$  and  $100^\circ\text{C}$  the hydrodynamic viscosity of water decreases by  $85\%$  from  $1.79 \text{ mN s m}^{-2}$  to  $0.27 \text{ mN s m}^{-2}$  whilst the absolute temperature only increases by  $37\%$  from  $273 \text{ K}$  to  $373 \text{ K}$ . Therefore, even though the SWCNTs diffuse slightly faster, the significantly lower viscosities allow them to propagate far further for a given photonic momentum and thus to separate faster.

Fig. 5 shows the variable space assessment of the effect of RPI and temperature on  $2\sigma$  separation times and found that an RPI of  $1.75$  at  $100^\circ\text{C}$  results in the shortest separation time of  $\sim 9$  minutes. This is  $12\times$  improvement over the  $\sim 109$  minute result for an RPI of  $1.0$  at room temperature ( $20^\circ\text{C}$ ) shown in the bottom of Fig. 3b. As discussed above, the elevated temperature for optimum separation can be explained due to the decreasing viscosity with increased temperature. However, the optimum RPI of  $1.75$  is more surprising. Once again, one would expect the shortest separation times to occur at an RPI of  $1.0$  where the photonic intensity is greatest without saturating the absorption band. While RPIs above  $1.0$  lead to faster separation because of the increased photonic intensity, the associated saturation causes slower separation because of the decreasing chiral discrimination. An optimum RPI above  $1.0$  suggests that there is a balance between the faster separation due to the increased intensity and the slower separation due to the increased saturation. Our results show that this balance occurs at an RPI of  $1.75$ .

Finally we fit a 2nd order polynomial surface to our variable space simulation results. This approach is chosen to impartially identify optimum separation conditions considering all simulations results. If no optimum conditions exist then a 2nd order polynomial fit has sufficient freedom to reduce to a linear fit and merely indicate a general trend. However, if optimum conditions do exist then these conditions are found at the surface minima. Any results that did not achieve  $2\sigma$  separation in the  $180$  minute simulated period are excluded prior to fitting.

The resulting 2nd order polynomial surface that predicts  $2\sigma$ -separation times as a function of temperature and RPI is shown in Fig. 5. A surface minimum is observed at an RPI of  $1.8$  and a temperature of  $100^\circ\text{C}$ . While this agrees well with the previously observed conditions for optimum separation (RPI:  $1.75$ , temperature:  $100^\circ\text{C}$ ) it predicts a near zero optimum  $2\sigma$  separation time. With respect to a  $\pm 12$  minute fitting error this is within acceptable limits and agrees with our observation that  $2\sigma$  separation occurs after  $9$  minutes under optimum conditions. We also take  $\pm 12$  minutes to be the error on our observed  $2\sigma$  separation times.



The  $\pm 12$  minute uncertainty is utilized to identify a suggested experimental operating region. This is indicated by the white contour line in Fig. 5. By studying the region constrained within this contour we suggest that elevated temperatures (for example above 40 °C) at an RPI of 1.8 should be initially investigated. An RPI of 1.8 corresponds to a photonic intensity of  $9 \times 10^7 \text{ W m}^{-2}$ . However, the maximum temperature that would be of use experimentally would have to consider the temperature stability of the dispersal method. This temperature will be dependent on the dispersal agent and there will be a balance between the improved separation rates afforded by higher operating temperatures and the potential instability of the sample. This would be a parameter which would necessitate an experimental investigation.

Also one should consider the experimental implications of absorption by small SWCNT bundles that may have implications on the photophoretic sorting efficiency. As mentioned previously bundles that contain many SWCNTs do not typically exhibit absorption bands whereas isolated SWCNTs do. However, it has been shown<sup>51,52</sup> *via* Exciton Energy Transfer (EET) that bundles that contain a small number of SWCNTs can exhibit preferential resonant absorption. Such bundles preferentially absorb at the resonant energy wavelength due to the presence of an on-resonance SWCNT in the bundle. The energy then couples into an adjacent SWCNT within the bundle before decaying either radiatively or non-radiatively. Fortunately, the energy transfer and potential re-emission at a lower energy should not have an effect on the sorting efficiency because the averaged re-emission is omni-directional and photophoresis is an absorption driven process. However, the off-resonance SWCNTs in the bundle would be extracted along with the preferentially absorbing on-resonance SWCNT and thus could contaminate the refined population. Such bundles will have a larger hydrodynamic radius than isolated SWCNTs and therefore propagate more slowly. We hope that the larger hydrodynamic radius (and hence frictional coefficient) will therefore limit the photonic drift velocities and mitigate the problems of small bundle absorption. We have not yet simulated it but we will conduct these studies soon.

It is also worth considering alternative experimental configurations as our simulation only considers a simple linear channel as an initial approach. However, the continuous introduction and extraction of SWCNT samples could compromise the sorting quality; for example in the device discussed here, all chiralities would travel in the same direction and are separated because they propagate at different speeds. Whilst it would be possible to use a photophoretic process to sort a single sample it would be more difficult to add a second sample. If a second sample were to be added at the start of the channel then the on-resonance SWCNTs of the second sample *could* catch up and mix with the off-resonance SWCNTs separated from the first sample. However, this may be easily overcome by implementing timed introduction and extraction cycles.

With this in mind, future work will simulate modified devices that utilize counter-propagation forces so that the separated SWCNT populations propagate in opposing directions. If the

magnitude of a counter-propagating force could be correctly balanced relative to the (propagating) photonic pressure forces then it could be possible to 'drive' the two populations to opposite ends of the sample chamber. Two such counter-propagating forces have been identified: fluidic counter-flow & optical counter-propagation. Fluidic counter-flow simulations will be used to investigate the fluidic conditions required for flowing the SWCNT solution towards the resonant optical source. Simulations of optical counter-propagation will investigate the use of an off-resonance optical source which acts equally upon all chiralities.

## Conclusion

Although procedures for chirality sorting SWCNT have emerged in recent years they are unable to guarantee the opto-electronic properties of the sorted samples. We have proposed using a resonance based optical sorting technique that by its very nature will guarantee the opto-electronic properties. To this end we have introduced and modeled the concept of photophoretic chiral sorting of SWCNT to investigate the effect of orientation, temperature and optical power on separation times. Whilst fixing the SWCNT orientation reduces the room temperature separation times from  $109 \pm 12$  minutes to  $84 \pm 12$  minutes, elevated temperatures and powers offer far greater enhancements. The  $2\sigma$  separation time can be reduced to less than 12 minutes by operating at  $9 \times 10^7 \text{ W m}^{-2}$  ( $1.8\times$  the saturation intensity) at 100 °C. These simulations show that optical sorting of SWCNTs is not only feasible but could be over 100 times faster than current approaches whilst guaranteeing high quality SWCNT separation. Although we appreciate the pressing need for experimental verification we propose that this work suggests an exciting avenue for chiral sorting of SWCNTs.

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## Notes and references

- 1 S. Iijima, *Nature*, 1991, **354**, 56–58.
- 2 C. X. Chen, W. Zhang, E. S. W. Kong and Y. F. Zhang, *Appl. Phys. Lett.*, 2009, **94**, 3.
- 3 Y. Jia, A. Y. Cao, X. Bai, Z. Li, L. H. Zhang, N. Guo, J. Q. Wei, K. L. Wang, H. W. Zhu, D. H. Wu and P. M. Ajayan, *Nano Lett.*, 2011, **11**, 1901–1905.
- 4 D. D. Tune, B. S. Flavel, R. Krupke and J. G. Shapter, *Adv. Energy Mater.*, 2012, **2**, 1043–1055.
- 5 A. R. Rocha, M. Rossi, A. Fazzio and A. J. R. da Silva, *Phys. Rev. Lett.*, 2008, **100**, 176083.
- 6 T. Ueda, M. M. H. Bhulyan, H. Norimatsu, S. Katsuki, T. Ikegami and F. Mitsugi, *Phys. E*, 2008, **40**, 2272–2277.
- 7 Z. Zanolli, R. Leghrib, A. Felten, J. J. Pireaux, E. Llobet and J. C. Charlier, *ACS Nano*, 2011, **5**, 4592–4599.
- 8 E. Poirier, R. Chahine, P. Benard, D. Cossement, L. Lafi, E. Melancon, T. K. Bose and S. Desilets, *Appl. Phys. A: Mater. Sci. Process.*, 2004, **78**, 961–967.



- 9 B. Panella, M. Hirscher and S. Roth, *Carbon*, 2005, **43**, 2209–2214.
- 10 K. Spyrou, D. Gournis and P. Rudolf, *ECS J. Solid State Sci. Technol.*, 2013, **2**, M3160–M3169.
- 11 P.-J. Tsai, C.-H. Yang, W.-C. Hsu, W.-T. Tsai and J.-K. Chang, *Int. J. Hydrogen Energy*, 2012, **37**, 6714–6720.
- 12 P. Avouris, Z. H. Chen and V. Perebeinos, *Nat. Nanotechnol.*, 2007, **2**, 605–615.
- 13 M. P. Anantram and F. Leonard, *Rep. Prog. Phys.*, 2006, **69**, 507–561.
- 14 M. L. Geier, P. L. Prabhumirashi, J. J. McMorro, W. C. Xu, J. W. T. Seo, K. Everaerts, C. H. Kim, T. J. Marks and M. C. Hersam, *Nano Lett.*, 2013, **13**, 4810–4814.
- 15 D. M. Sun, C. Liu, W. C. Ren and H. M. Cheng, *Small*, 2013, **9**, 1188–1205.
- 16 C. Wang, K. Takei, T. Takahashi and A. Javey, *Chem. Soc. Rev.*, 2013, **42**, 2592–2609.
- 17 M. M. Shulaker, G. Hills, N. Patil, H. Wei, H.-Y. Chen, H. S. P. Wong and S. Mitra, *Nature*, 2013, **501**, 526–530.
- 18 M. M. Shokrieh and R. Rafiee, *Mech. Compos. Mater.*, 2010, **46**, 155–172.
- 19 T. W. Odom, J. L. Huang, P. Kim and C. M. Lieber, *J. Phys. Chem. B*, 2000, **104**, 2794–2809.
- 20 A. Vijayaraghavan, *J. Mater. Chem.*, 2012, **22**, 7083–7087.
- 21 M. S. Arnold, A. A. Green, J. F. Hulvat, S. I. Stupp and M. C. Hersam, *Nat. Nanotechnol.*, 2006, **1**, 60–65.
- 22 A. A. Green and M. C. Hersam, *Adv. Mater.*, 2011, **23**, 2185–2190.
- 23 A. L. Antaris, J. W. T. Seo, A. A. Green and M. C. Hersam, *ACS Nano*, 2010, **4**, 4725–4732.
- 24 A. L. Antaris, J. W. T. Seo, R. E. Brock, J. E. Herriman, M. J. Born, A. A. Green and M. C. Hersam, *J. Phys. Chem. C*, 2012, **116**, 20103–20108.
- 25 J. A. Fagan, M. L. Becker, J. Chun and E. K. Hobbie, *Adv. Mater.*, 2008, **20**, 1609–1613.
- 26 A. A. Green, M. C. Duch and M. C. Hersam, *Nano Res.*, 2009, **2**, 69–77.
- 27 M. S. Arnold, S. I. Stupp and M. C. Hersam, *Nano Lett.*, 2005, **5**, 713–718.
- 28 S. Ghosh, S. M. Bachilo and R. B. Weisman, *Nat. Nanotechnol.*, 2010, **5**, 443–450.
- 29 N. Berton, F. Lemasson, J. Tittmann, N. Sturzl, F. Hennrich, M. M. Kappes and M. Mayor, *Chem. Mater.*, 2011, **23**, 2237–2249.
- 30 M. Kawai, H. Kyakuno, T. Suzuki, T. Igarashi, H. Suzuki, T. Okazaki, H. Kataura, Y. Maniwa and K. Yanagi, *J. Am. Chem. Soc.*, 2012, **134**, 9545–9548.
- 31 X. Tu, S. Manohar, A. Jagota and M. Zheng, *Nature*, 2009, **460**, 250–253.
- 32 H. P. Liu, D. Nishide, T. Tanaka and H. Kataura, *Nat. Commun.*, 2011, **2**, 8.
- 33 A. Nish, J. Y. Hwang, J. Doig and R. J. Nicholas, *Nat. Nanotechnol.*, 2007, **2**, 640–646.
- 34 L. Van Hove, *Phys. Rev.*, 1953, **89**, 1189–1193.
- 35 M. J. Mendes, H. K. Schmidt and M. Pasquali, *J. Phys. Chem. B*, 2008, **112**, 7467–7477.
- 36 O. M. Marago, P. H. Jones, F. Bonaccorso, V. Scardaci, P. G. Gucciardi, A. G. Rozhin and A. C. Ferrari, *Nano Lett.*, 2008, **8**, 3211–3216.
- 37 D. M. Carberry, S. H. Simpson, J. A. Grieve, Y. Wang, H. Schafer, M. Steinhart, R. Bowman, G. M. Gibson, M. J. Padgett, S. Hanna and M. J. Miles, *Nanotechnology*, 2010, **21**, 7.
- 38 D. B. Phillips, D. M. Carberry, S. H. Simpson, H. Schafer, M. Steinhart, R. Bowman, G. M. Gibson, M. J. Padgett, S. Hanna and M. J. Miles, *J. Opt.*, 2011, **13**, 8.
- 39 S. H. Simpson and S. Hanna, *J. Opt. Soc. Am. A*, 2011, **28**, 850–858.
- 40 S. H. Simpson and S. Hanna, *Nanotechnology*, 2012, **23**, 205502.
- 41 M. S. Arnold, J. E. Sharping, S. I. Stupp, P. Kumar and M. C. Hersam, *Nano Lett.*, 2003, **3**, 1549–1554.
- 42 Y. Q. Tan and D. E. Resasco, *J. Phys. Chem. B*, 2005, **109**, 14454–14460.
- 43 M. E. Itkis, D. E. Perea, S. Niyogi, S. M. Rickard, M. A. Hamon, B. Zhao and R. C. Haddon, *Nano Lett.*, 2003, **3**, 309–314.
- 44 M. J. O'Connell, S. M. Bachilo, C. B. Huffman, V. C. Moore, M. S. Strano, E. H. Haroz, K. L. Rialon, P. J. Boul, W. H. Noon, C. Kittrell, J. P. Ma, R. H. Hauge, R. B. Weisman and R. E. Smalley, *Science*, 2002, **297**, 593–596.
- 45 H. J. Keh and H. J. Tu, *Colloids Surf., A*, 2001, **176**, 213–223.
- 46 N. T. Tong, *J. Colloid Interface Sci.*, 1975, **51**, 143–151.
- 47 R. Kuhn and S. Hoffstetterkuhn, *Chromatographia*, 1992, **34**, 505–512.
- 48 F. Albertorio, M. E. Hughes, J. A. Golovchenko and D. Branton, *Nanotechnology*, 2009, **20**, 9.
- 49 D. A. Tsyboulski, S. M. Bachilo, A. B. Kolomeisky and R. B. Weisman, *ACS Nano*, 2008, **2**, 1770–1776.
- 50 M. M. Tirado, C. L. Martinez and J. G. Delatorre, *J. Chem. Phys.*, 1984, **81**, 2047–2052.
- 51 P. H. Tan, A. G. Rozhin, T. Hasan, P. Hu, V. Scardaci, W. I. Milne and A. C. Ferrari, *Phys. Rev. Lett.*, 2007, **99**, 4.
- 52 O. N. Torrens, D. E. Milkie, M. Zheng and J. M. Kikkawa, *Nano Lett.*, 2006, **6**, 2864–2867.

