

Chem Soc Rev

Chemical Society Reviews

rsc.li/chem-soc-rev



ISSN 0306-0012



Cite this: *Chem. Soc. Rev.*, 2025, 54, 2154

Optimizing photocatalysis *via* electron spin control

Shaoxiong He,^{ab} Yanxi Chen,^b Jingyun Fang,^b Yijiang Liu^{*ac} and Zhiqun Lin^{ab}

Solar-driven photocatalytic technology holds significant potential for addressing energy crisis and mitigating global warming, yet is limited by light absorption, charge separation, and surface reaction kinetics. The past several years has witnessed remarkable progress in optimizing photocatalysis *via* electron spin control. This approach enhances light absorption through energy band tuning, promotes charge separation by spin polarization, and improves surface reaction kinetics *via* strengthening surface interaction and increasing product selectivity. Nevertheless, the lack of a comprehensive and critical review on this topic is noteworthy. Herein, we provide a summary of the fundamentals of electron spin control and the techniques employed to scrutinize the electron spin state of active sites in photocatalysts. Subsequently, we highlight advanced strategies for manipulating electron spin, including doping design, defect engineering, magnetic field regulation, metal coordination modulation, chiral-induced spin selectivity, and combined strategies. Additionally, we review electron spin control-optimized photocatalytic processes, including photocatalytic water splitting, CO₂ reduction, pollutant degradation, and N₂ fixation, providing specific examples and detailed discussion on underlying mechanisms. Finally, we outline perspectives on further enhancing photocatalytic activity through electron spin manipulation. This review seeks to offer valuable insights to guide future research on electron spin control for improving photocatalytic applications.

Received 27th September 2024

DOI: 10.1039/d4cs00317a

rsc.li/chem-soc-rev

^a Department of Chemical and Biomolecular Engineering, National University of Singapore, 4 Engineering Drive 4, Singapore 117585, Singapore.

E-mail: z.lin@nus.edu.sg

^b School of Environmental Science and Engineering, Guangdong Provincial Key Laboratory of Environmental Pollution Control and Remediation Technology, Sun Yat-sen University, Guangzhou 510275, P. R. China.

E-mail: fangy3@mail.sysu.edu.cn

^c College of Chemistry, Key Lab of Environment-Friendly Chemistry and Application in Ministry of Education, Xiangtan University, Xiangtan 411105, Hunan Province, P. R. China. E-mail: liuyijiang84@xtu.edu.cn



Shaoxiong He

Shaoxiong He is a PhD candidate in Professor Jingyun Fang's group in the School of Environmental Science and Engineering at Sun Yat-sen University, China. He is also a visiting student in Professor Zhiqun Lin's group in the Department of Chemical and Biomolecular Engineering at the National University of Singapore. His research focuses on photocatalyst design for energy conversion and environmental remediation.



Yanxi Chen

Yanxi Chen is a PhD candidate in the School of Environmental Science and Engineering at Sun Yat-sen University, China. Her research focuses on photocatalyst design and microbiome engineering for environmental remediation.



energy received by the Earth in just one hour (4.3×10^{20} J) is equivalent to the entire global energy consumption for an entire year (4.1×10^{20} J).¹ Forecasts predict that by 2035, over 50% of global power generation will be derived from renewable sources, predominantly solar energy, highlighting the crucial role of solar energy conversion technology in future sustainable development.² Solar-driven photocatalytic technology has been extensively researched to facilitate water splitting for green hydrogen production and convert CO_2 into solar fuels (e.g., CO , CH_4 , CH_3OH , and $\text{C}_2\text{H}_5\text{OH}$), offering an effective solution to both energy and environmental crises.^{3–6} Despite significant achievements in photocatalysis over recent decades – such as doping design, co-catalyst introduction, defect engineering, heterojunction construction, and external field modulation – its efficiency remains constrained by challenges including limited light absorption, inefficient charge separation, and insufficient surface reaction kinetics.^{7–9} Thus, developing a



Jingyun Fang

Dr Jingyun Fang is a Professor in the School of Environmental Science and Engineering at Sun Yat-sen University, China. She received her bachelor and doctoral degrees from Harbin Institute of Technology in 2003 and 2010, respectively. She was a postdoctoral fellow at the Hong Kong University of Science & Technology between 2010 and 2012. Her research focuses on water chemistry, environmental functional materials, and physico-chemical water treatment technologies including oxidation/reduction processes, and formation & control of disinfection byproducts.



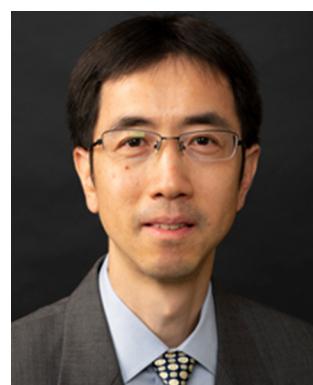
Yijiang Liu

Dr Yijiang Liu is a Professor in the College of Chemistry at Xiangtan University, China. She received her MS degree from Xiangtan University in 2009 and PhD degree from the Institute of Chemistry, Chinese Academy of Sciences in 2015. She was a visiting scholar at Georgia Institute of Technology during 2017–2018. Her research work focuses on design and preparation of advanced porous carbon materials as efficient and durable

electrocatalysts for energy conversion and storage; crafting of highly stable perovskite nanocrystals via functional polymer nanoreactors; and Janus materials for concurrent catalysis and emulsification.

universal strategy to overcome these limitations and achieve highly efficient solar energy conversion is of paramount importance.

Recently, electron spin control has emerged as a promising strategy for enhancing photocatalytic performance.^{10–13} Electron spin control involves the manipulation of both electron spin and spin states. Electron spin refers to the intrinsic angular momentum of an electron, characterized by spin-up or spin-down configurations. In contrast, the electron spin state describes the overall spin arrangement of electrons within an atomic orbital and can be classified into singlet, doublet, triplet, and other multiplet states.^{14,15} These two concepts are fundamental intrinsic characteristics that predominantly govern the physical properties and chemical behaviors of photocatalysts, offering innovative approaches to enhance photocatalytic efficiency.^{12–16} For example, electron spin control can be used to tune the energy band structures of photocatalysts, extending their light absorption capabilities across a broader range of solar spectra, including visible and near-infrared (NIR) light.^{17–19} Moreover, spin polarization can promote the separation efficiency of photogenerated electrons and holes, accelerating charge migration to the photocatalyst surface.^{13,20–23} Additionally, the interaction between photocatalysts and reactants is strengthened by modulating the electron spin state of active sites, improving surface reaction kinetics.^{24–26} Notably, aligning electron spin in specific directions has been proven to enhance the selectivity for desired products, offering a novel strategy for regulating surface reaction pathways. For example, this approach favors the production of triplet O_2 over singlet hydrogen peroxide (H_2O_2) in photocatalytic water splitting processes.^{12,13,26–29} Therefore, electron spin control holds great potential for broadening light absorption range, mitigating charge recombination, and boosting the efficiency and selectivity of surface reactions in photocatalysis. Despite the growing interest in electron spin manipulation strategies for boosting photocatalytic activity over the past decade, a comprehensive



Zhiqun Lin

Dr Zhiqun Lin is a Professor in the Department of Chemical and Biomolecular Engineering at the National University of Singapore. His research interests include photocatalysis, solar energy conversion, electrocatalysis, batteries, semiconductor organic–inorganic nanocomposites, multifunctional nanocrystals, conjugated polymers, block copolymers, hierarchical structure formation and assembly, and surface and interfacial properties.



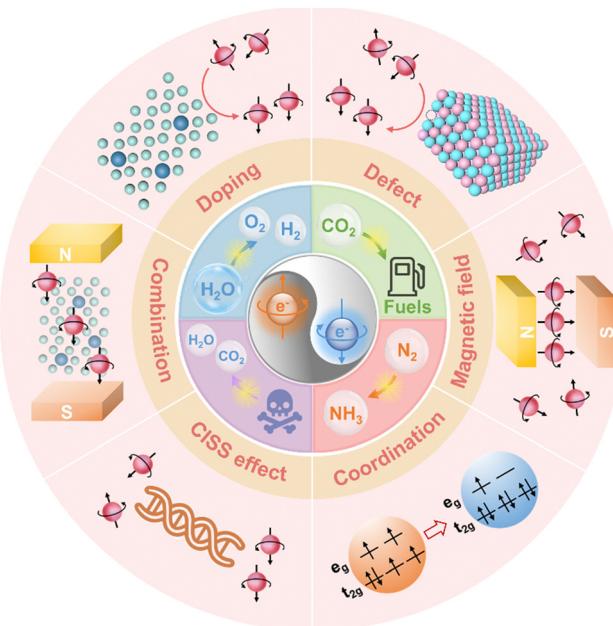


Fig. 1 Overview of the mechanisms, strategies, and photocatalytic applications of electron spin control.

and critical review summarizing the mechanisms, strategies, and applications of electron spin control in photocatalysis remains notably absent.

In this review, we aim at providing a systematic overview of how electron spin control can optimize photocatalysis, the strategies and materials involved, and their applications in various photocatalytic processes (Fig. 1). We begin with the fundamentals of photocatalysis and electron spin control, followed by exploration of the role of electron spin control in photocatalysis. Subsequently, we highlight advanced strategies for manipulating electron spin, including doping design, defect engineering, magnetic field regulation, metal coordination modulation, chiral-induced spin selectivity, and combined approaches. We also discuss recent developments in these strategies as applied to photocatalytic water splitting, CO_2 reduction, pollutant degradation, and N_2 fixation. Finally, we outline

current challenges and future research directions in this rapidly evolving field.

2. Fundamentals and challenges of photocatalysis

2.1 Fundamentals of photocatalysis

Fig. 2 depicts the detailed steps of photocatalysis in a photocatalyst, from light excitation to surface redox reaction. First, when a photocatalyst is exposed to light with energy equal to or greater than its band gap (step 1), photogenerated electrons are excited from the valence band (VB) to the conduction band (CB), leaving holes in the VB (step 2). Some electron–hole pairs may return to the ground state *via* radiative or non-radiative relaxation, typically occurring on a nanosecond to picosecond time scale, which hinders the formation of free charge carriers. Some electrons and holes can overcome the exciton binding energy and separate into free charge carriers (step 3). However, these charge carriers may recombine within the bulk of photocatalyst (step 4), but if their diffusion length is sufficient, they can reach the surface of the photocatalyst (step 5). The diffusion of charge carriers typically happens on a picosecond time scale and requires a concentration gradient to drive their movement. Unfortunately, some charge carriers may recombine before they can participate in surface reactions (step 6). Ultimately, the photocatalytic process concludes as the remaining charge carriers are consumed in surface reactions (steps 7 and 8), which generally occur on a longer time scale. In the surface reactions, electron-driven reduction enables H_2 production and the conversion of CO_2 into solar fuels, N_2 into NH_3 or NH_4^+ , and O_2 into superoxide radical (O_2^\cdot) or H_2O_2 . Meanwhile, hole-driven oxidation can convert H_2O into hydroxyl radicals (HO^\cdot), H_2O_2 , or O_2 .^{9,30–32}

Notably, the surface redox reactions can only be effectively driven by photogenerated electrons and holes when the redox potentials lie between the CB and VB potentials of a photocatalyst. For instance, to achieve overall water splitting, the CB must be more negative than the reduction potential for H_2 production, while the VB must be more positive than the

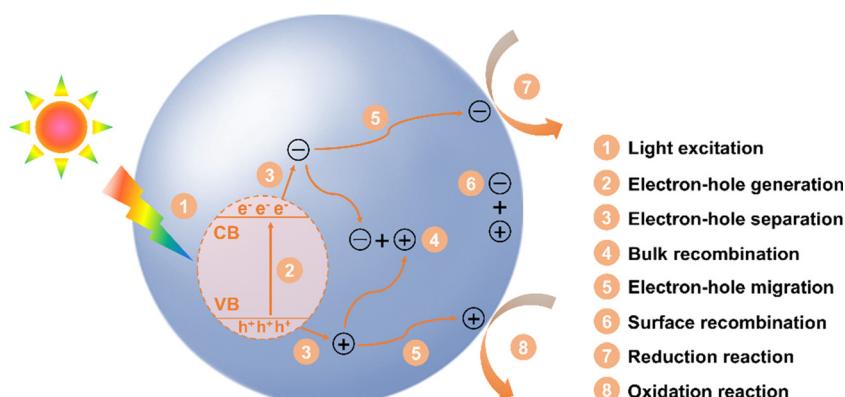


Fig. 2 Schematic diagram of photocatalytic mechanism.



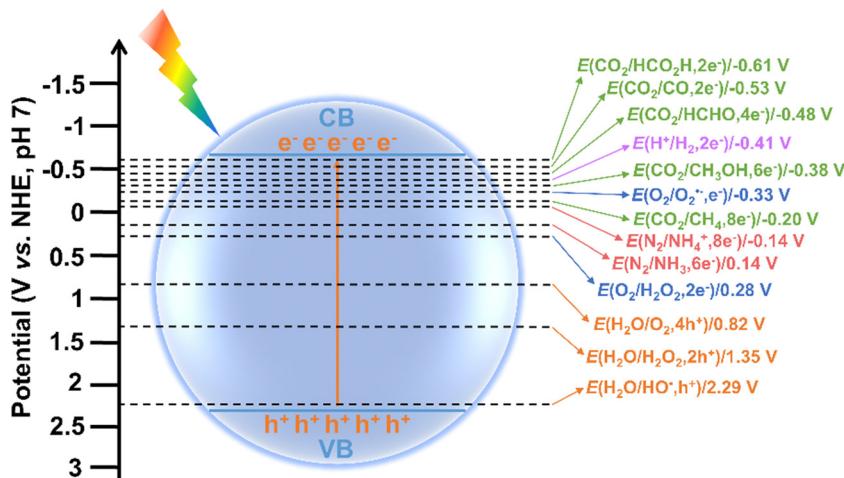


Fig. 3 Redox potentials of different species in photocatalysis.

oxidation potential required for O_2 generation. The redox potentials of different species in photocatalysis are shown in Fig. 3. It is accepted that a more negative CB position of a photocatalyst enhances its ability to drive reduction reactions, while a more positive VB position favors oxidation reactions. Interestingly, multi-electron reduction of O_2 and CO_2 are thermodynamically more favorable than their single-electron counterparts due to their lower reduction potentials. Similarly, the four-electron oxidation of H_2O is more readily achieved than the two-electron and single-electron oxidations, as the former has a higher oxidation potential. Therefore, whether a photocatalytic reaction can occur and its efficiency largely depend on the CB and VB positions of photocatalysts in relation to the redox potentials of target reactions. This provides guidance for selecting appropriate photocatalysts for specific target reactions. For example, TiO_2 , with a CB potential of -0.5 V and a VB potential of 2.7 V, is a commonly used photocatalyst for water splitting due to its suitable band potentials, stable chemical property, and low cost.³³ Typically, photocatalysts with a larger band gap have VB and CB potentials that are more suitable for overall water splitting, but this often results in reduced light absorption capability. The absorption spectrum of TiO_2 is confined to the ultraviolet (UV) range (about 4% of sunlight), limiting its visible light utilization efficiency. Studies have shown that doping TiO_2 with metal or non-metal elements (e.g., N, S, Fe, and Co), can extend its light absorption into the visible region.³³ In contrast, $\text{g-C}_3\text{N}_4$ and CdS , which have smaller band gaps, can absorb visible light and possess CB potentials of approximately -1.1 V and -0.8 V, respectively.^{34,35} This makes them more effective for H_2 generation compared to TiO_2 . However, their VB potentials are inadequate for water oxidation, so a co-catalyst or a heterojunction with other suitable semiconductors is needed to achieve overall water splitting. Thus, it is essential to carefully balance and optimize the trade-off between redox potential (determined by CB and VB potentials) and light absorption capability (governed by band gap). Additionally, for certain specific reactions, the selection of photocatalyst can be based primarily on its VB or CB potential.

For instance, in CO_2 reduction, a photocatalyst with a relatively negative CB is ideal, as it enhances the reduction process irrespective of the VB potential.⁹ On the contrary, for the degradation of organic pollutants, photocatalysts with a relatively positive VB potential is preferable. This is because they possess stronger hole oxidation capability, allowing them to generate highly reactive HO^\bullet , which significantly improves the efficiency of organic pollutant degradation.

The overall efficiency of photocatalysis is primarily influenced by the combined effects of three key steps: light absorption, charge separation, and surface reaction kinetics. Consequently, the solar energy conversion efficiency can be determined by eqn (1).

$$\eta = \eta_{\text{abs}} \times \eta_{\text{cs}} \times \eta_{\text{sr}} \quad (1)$$

where η is the solar energy conversion efficiency, η_{abs} represents the light absorption efficiency, η_{cs} denotes the charge separation efficiency, and η_{sr} signifies the surface reaction efficiency.

Quantum efficiency (QE) is another key metric for evaluating photocatalytic reactions. It quantifies the rate at which a specific event occurs per absorbed photon and is used to assess the efficiency of reactant consumption, product formation, and light emission in photophysical and photochemical reactions.³⁶ In photocatalysis, a higher QE indicates that more photons are effectively utilized in the reaction, suggesting a more efficient photocatalyst. Optimizing QE can lead to improved photocatalyst design and enhanced performance in photocatalytic reactions. The actual QE, often referred to as internal quantum efficiency (IQE), is defined as the number of reacted electrons participating in the reaction to the total number of photons absorbed over a given period.² However, due to challenges such as light scattering and reflection during photocatalysis, directly measuring the number of photons absorbed is difficult. As a result, apparent quantum efficiency (AQE) is typically used as an alternative. AQE is determined by the number of reacted electrons to the number of incident photons. Common detectors used to measure incident photons include thermopiles and silicon photodiodes, which

can accurately gauge the flux of incident photons. Since light intensity within the irradiated area is often uneven, it is advisable to calculate the photon number by integrating the photon flux across the entire illuminated area. This involves dividing the illuminated area into smaller segments, measuring the photon flux in each segment, and then integrating these measurements to obtain an average photon number. The IQE and AQE of a photocatalyst can be determined by eqn (2) and (3).

$$\text{IQE} = \frac{\text{number of reacted electrons}}{\text{number of absorbed photons}} \times 100\% \quad (2)$$

$$\text{AQE} = \frac{\text{number of reacted electrons}}{\text{number of incident photons}} \times 100\% \quad (3)$$

2.2 Photocatalytic applications

Photocatalytic technology holds tremendous promise in the fields of modern energy and environmental science, particularly in key processes such as H₂ production *via* water splitting, CO₂ reduction, pollutant degradation, and N₂ fixation. By harnessing light energy to drive chemical reactions, photocatalysis not only enables the efficient generation of renewable energy but also offers a low-energy, environmentally friendly solution for reducing greenhouse gas emissions and addressing environmental pollution. In water splitting, photocatalysts can directly decompose water into H₂ and O₂, providing a clean energy source; in CO₂ reduction, they transform greenhouse gases into valuable fuels and chemicals, contributing to the development of a carbon-recycling economy. Additionally, in pollutant degradation and N₂ fixation, photocatalytic technology plays a crucial role in environmental remediation and agricultural production. Due to its sustainability and versatility, photocatalysis has emerged as a promising solution to global energy and environmental challenges. As photocatalysts and reaction mechanisms continue to improve, the technology is poised to make an even greater impact in large-scale industrial applications.

2.2.1 Photocatalytic water splitting. Photocatalytic water splitting is a process that converts light energy into H₂ and O₂. The overall reaction consists of two half-reactions: water oxidation (2H₂O → O₂ + 4H⁺ + 4e⁻) and water reduction (4H⁺ + 4e⁻ → 2H₂), driven by photogenerated holes and electrons, respectively. For efficient photocatalytic water splitting, photocatalysts must meet several key requirements, including an appropriate band gap, suitable conduction and valence band positions to enable both reduction and oxidation, and high stability and conductivity. Additionally, their ability to respond to visible light is crucial for enhancing photocatalytic efficiency. Commonly used photocatalysts primarily include semiconductor oxides, sulfides, and nitrides. Among them, TiO₂ is one of the most extensively studied materials, known for its excellent stability and strong oxidative properties. However, TiO₂ has a relatively large band gap (about 3.2 eV), which limits its light absorption capability. In contrast, photocatalysts like g-C₃N₄, CdS, and BiVO₄, with their smaller band gaps, are responsive to

visible light.³⁰ Nevertheless, various modification strategies, such as doping, precious metal loading, heterostructure construction, and defect engineering, are needed to enhance the separation of photogenerated electrons and holes, thereby improving photocatalytic efficiency.²

In the photocatalytic H₂ production half-reaction, research focuses on optimizing photocatalysts and reaction conditions to improve the H₂ production rate and quantum efficiency. Studies have shown that introducing precious metal nanoparticles with surface plasmon resonance effects, such as Pt and Au, can enhance light absorption and improve the separation efficiency of photogenerated charge carriers.² Additionally, regulating the crystal face structure of the photocatalyst, such as exposing specific crystal facets, can increase the selectivity and efficiency of the H₂ production reaction.³⁷

The main challenge in the photocatalytic O₂ production half-reaction lies in the high overpotential and slow reaction kinetics of the water oxidation reaction. To address this, researchers have introduced highly active oxidation catalysts, such as Co₃O₄ and IrO₂, to accelerate the water oxidation process.³⁸ Moreover, constructing 3D nanostructured materials, such as BiVO₄ photoanodes, has increased the contact area between light absorption sites and the reaction interface, thereby enhancing water oxidation efficiency.³⁹

Despite advancements, photocatalytic overall water splitting still faces significant challenges, particularly in simultaneously improving the efficiency of both H₂ and O₂ production reactions. Strategies like constructing Z-scheme heterojunctions and photo-alloy catalysts have been shown to promote the overall water splitting.³⁰ Additionally, utilizing two-photon excitation and multi-electron transfer mechanisms has improved the quantum efficiency of overall water splitting.² As research deepens our understanding of photocatalytic mechanisms and photocatalysts continue to be optimized, photocatalytic water splitting holds great potential for practical applications in the future.

2.2.2 Photocatalytic CO₂ reduction. Photocatalytic CO₂ reduction is a technology that harnesses light energy to convert CO₂ into valuable chemicals. This process primarily relies on photocatalysts that absorb photons and generate electrons to drive the CO₂ reduction reaction. Such a process must satisfy thermodynamic requirements; namely, the redox potential of the reduction reaction must be more positive than the CB of the photocatalyst. Additionally, from the perspective of reaction kinetics, it is essential to have catalytic sites available for the activation of CO₂. Common reduction products include carbon monoxide (CO), methane (CH₄), formic acid (HCOOH), formaldehyde (HCHO), methanol (CH₃OH) and various long-chain carbon compounds.

A crucial aspect of optimizing the efficiency and selectivity is adjusting both the reaction conditions and the characteristics of the photocatalyst. Research has demonstrated that the reaction efficiency and the selectivity for different products can be influenced by adjusting reaction conditions, such as temperature, pressure, CO₂ concentration, and light intensity.⁵ Moreover, the fabrication of heterostructured photocatalysts with well-aligned energy bands is an effective strategy for



achieving a broad spectral response and efficient charge separation.³⁰ Additionally, photocatalytic CO₂ reduction can be promoted using co-catalysts such as Au, Ag, Cu, and Bi, primarily due to the localized surface plasmon resonance effect.⁴⁰ Notably, by altering the compositions, morphologies, and crystal structures of photocatalysts, their electronic structure can be modified, which in turn enhances its selectivity for specific products.⁵ With a deeper understanding of the reaction mechanisms and the development of novel photocatalysts, it is anticipated that more precise control over the products of photocatalytic CO₂ reduction will be achieved in the future.

2.2.3 Photocatalytic pollutant degradation. Photocatalytic technology has significant potential for environmental remediation, primarily due to its ability to effectively decompose organic pollutants, thereby purifying water and air. The mechanism behind photocatalytic pollutant degradation involves a series of complex chemical reactions, with the generation of free radicals and their interactions with pollutants being crucial. Under appropriate thermodynamic conditions, when the VB and CB positions of photocatalysts are suitable, holes can react with H₂O to produce HO[•], while electrons react with O₂ to generate O₂^{•-}.⁴¹ These free radicals effectively degrade organic pollutants into harmless products or even fully mineralize them into CO₂. Additionally, electrons can produce H₂O₂ through a two-electron reduction pathway or generate HO[•] *via* a three-electron reduction pathway.⁴² Although holes possess oxidizing abilities, their effectiveness is typically determined by the VB positions of photocatalysts.⁴³

Currently, many researchers utilize photocatalysts in combination with common oxidants, such as H₂O₂, O₃, persulfate, and hypochlorous acid.⁴⁴⁻⁴⁷ The reactions between electrons or holes and these oxidants generate additional free radicals, significantly enhancing pollutant degradation efficiency. Therefore, photocatalytic pollutant degradation technology offers a promising solution for environmental remediation.

2.2.4 Photocatalytic N₂ fixation. Photocatalytic N₂ fixation offers a green and efficient method for sustainable nitrogen fertilizer production. It harnesses solar energy to directly convert atmospheric N₂ into NH₃, providing an alternative to the traditional, energy-intensive Haber–Bosch process. Photocatalytic N₂ fixation operates under mild conditions, without the need for high temperatures and pressures, reducing reliance on fossil fuels and lowering both energy consumption and carbon emissions. During photocatalytic N₂ fixation for NH₃ production, the two nitrogen atoms of N₂ remain connected. The reaction proceeds *via* two main pathways: in the distal pathway, the nitrogen atoms farther from the photocatalyst surface are hydrogenated first, followed by the nitrogen atoms adsorbed on the surface until ammonia is formed; in the alternating pathway, the two nitrogen atoms are hydrogenated alternately until the process is complete.⁴⁸ These pathways influence both the efficiency of N₂ fixation and product selectivity.

Recent advancements in photocatalytic N₂ fixation focus on the development of new photocatalysts, the optimization of surface active sites, and the enhancement of electron transfer efficiency. To address challenges related to N₂ adsorption and

high activation energy barriers, researchers have employed strategies such as morphology control, element doping, defect engineering, and heterojunction design.⁴⁸⁻⁵⁰ These approaches improve N₂ adsorption and activation while reducing the recombination rate of photogenerated charge carriers. These innovations lay the groundwork for achieving efficient NH₃ production at room temperature and pressure, paving the way for its large-scale application.

2.3 Current challenges in photocatalysis

Despite considerable progress in photocatalysis over the past few decades, its efficiency remains hindered by challenges including limited light absorption, inefficient charge separation, and insufficient surface reaction kinetics.

2.3.1 Limited light absorption. Light absorption is the crucial first step in photocatalysis, directly influencing the generation of photogenerated electron–hole pairs and, therefore, determining the overall performance of photocatalysts. The solar spectrum consists of UV light (4%), visible light (43%), and NIR light (53%). Traditional photocatalysts, such as TiO₂, have a large band gap (~3.2 eV) and can only absorb UV light with wavelengths below 400 nm. This limits their ability to utilize only 4% of sunlight, leaving the majority of solar energy untapped. As a result, expanding the absorption range to visible and NIR regions has become a critical focus for improving photocatalytic efficiency.

Researchers have explored various strategies for adjusting the band gap of photocatalysts to enhance light absorption. One common approach is doping, where transition metals (*e.g.*, Fe, Co, and Mn) and non-metals (*e.g.*, N, S, and C) are introduced into photocatalysts.³⁰ This can reduce the band gap and extend the absorption range to visible and NIR regions. However, doping may introduce impurity states that increase the recombination rate of photogenerated charge carriers, potentially reducing quantum efficiency. Another method to extend the absorption range is sensitization with organic dyes and quantum dots.^{51,52} However, the long-term stability of these sensitizers and their susceptibility to photodegradation under extended illumination remain challenges. Additionally, some narrow bandgap materials, such as black phosphorus,⁵³ Ag₂S,⁵⁴ and ZnIn₂S₄,⁵⁵ have been developed to absorb visible and even NIR light. Despite this, these materials are often vulnerable to photocorrosion and rapid recombination of electrons and holes, leading to reduced activity and poor long-term stability. Continued optimization is needed to achieve both efficient and stable visible-to-NIR solar energy conversion.

2.3.2 Inefficient charge separation. The separation of photogenerated charge carriers is a critical step in photocatalysis, as it directly influences both QE and overall reaction efficiency. However, photogenerated electrons and holes tend to recombine within photocatalyst and at its surface, especially in the presence of lattice defects and abundant surface states. This recombination leads to the loss of light energy and significantly reduces photocatalytic efficiency.

To address this issue, researchers have developed several strategies with notable progress. One effective approach is



constructing semiconductor heterojunctions with appropriately matched energy band structures.³⁰ This creates an interface that forms an electric field, which facilitates the migration of electrons and holes to different semiconductors, thereby reducing their recombination and enhancing photocatalytic efficiency. For example, the TiO₂/g-C₃N₄ heterojunction combines TiO₂'s high photocatalytic activity with g-C₃N₄'s visible light absorption capability, utilizing the built-in electric field to achieve improved charge separation.^{56,57} However, designing heterojunctions requires precise band structure alignment between the semiconductors, which limits the range of materials that can be used. Moreover, many materials face challenges in band structure matching and lattice compatibility, reducing the diversity and flexibility of heterojunction photocatalysts.

Introducing suitable co-catalysts (*e.g.*, Pt, Pd, RuO₂, and Co₃O₄) is another widely studied approach to promote charge separation.^{2,30} These co-catalysts facilitate the transport of electrons or holes, thereby boosting the overall photocatalytic performance. For instance, Pt and RuO₂ are commonly used in photocatalytic water splitting to enhance H₂ and O₂ production, respectively, by serving as sinks for electrons and holes.⁵⁸ However, efficient co-catalysts, particularly precious metals such as Pt, Au, and Pd, are costly, limiting their large-scale use. Additionally, the chemical stability of these co-catalysts may diminish over time, posing challenges for long-term photocatalytic applications. Therefore, there remains a need to develop efficient, cost-effective, and versatile strategies to facilitate charge separation.

2.3.3 Insufficient surface reaction kinetics. The final step in photocatalysis is the surface reactions on photocatalysts, where photogenerated electrons and holes participate in chemical transformations to produce useful products or achieve specific reactions. Key factors affecting surface reaction kinetics include reactant adsorption, reaction energy barrier, and product desorption. Specifically, it is crucial for reactants to be effectively adsorbed onto photocatalyst surface to strengthen their interaction and facilitate reactions. Additionally, thermodynamically favorable reaction pathways with low energy barriers are desirable. Moreover, products should be desorbed rapidly to prevent their accumulation on the catalyst surface.

Current approaches for improving surface reaction kinetics focus on increasing the number of active sites on photocatalyst surface and enhancing the adsorption of reactants and intermediates. For example, compared to bulk materials, the design of nanomaterials (*e.g.*, nanosheets, nanowires, and nanospheres) increases the specific surface area and provides more active sites.⁵⁹ Additionally, the introduction of co-catalysts can increase active sites and lower reaction energy barriers, facilitating better reactant adsorption and more efficient reaction pathways with reduced energy barriers.⁶⁰ Creating defects within photocatalyst structure can also offer extra reaction sites and enhance the adsorption of reactants.⁶¹ Furthermore, functionalizing photocatalyst surface with suitable ligands (*e.g.*, amino acids, alcohols, and organic acids)

strengthens the interaction between the photocatalysts and target molecules, thereby boosting surface reaction kinetics.⁶²

Notably, selective product generation, where catalysts preferentially produce desired products while minimizing the formation of byproducts, is crucial for maximizing surface reaction efficiency. However, common photocatalysts may lack high selectivity, leading to the formation of various byproducts, which reduces the yields of target products and complicates purification. For instance, the selectivity for O₂ production from H₂O is often unsatisfied, as it tends to produce H₂O₂ as a byproduct, making this the rate-limiting step in photocatalytic water splitting.¹² Similarly, in the photocatalytic degradation of organic pollutants, photocatalysts may generate toxic intermediates and byproducts such as phenols and aldehydes.^{45,63} These byproducts not only reduce the overall effectiveness but also pose additional environmental risks. Therefore, it is essential to develop highly selective photocatalysts to optimize surface reactions.

While photocatalysis is hindered by limited light absorption, inefficient charge separation, and insufficient surface reaction kinetics, recent research has proposed innovative solutions to overcome these challenges. Among these, electron spin control has emerged as a particularly promising method. By manipulating the spin state of electrons within photocatalysts, electron spin control can optimize photocatalysis through enhancing light absorption, promoting charge separation, and improving surface reaction kinetics. In the following sections, we will summarize the fundamentals of electron spin control and its role in photocatalysis.

3. Fundamentals of electron spin control and its role in photocatalysis

3.1 Theories and principles of electron spin and spin state

In this review, electron spin control encompasses the manipulation of both electron spin and spin states. Electron spin refers to the intrinsic angular momentum of an electron, characterized by the spin magnetic quantum number m_s , which can be either spin up ($m_s = \frac{1}{2}$) or spin down ($m_s = -\frac{1}{2}$) (Fig. 4a); while electron spin state describes the overall spin configuration of electrons in an atomic orbital and can be classified as singlet, doublet, triplet, *etc.*^{14,15} These fundamental concepts significantly impacts the physical properties and chemical behavior of atoms, molecules, and materials.

The theories and principles of electron spin and spin state have been developed over the past century, establishing a solid foundation for research related to electron spin control. In 1925, Wolfgang Pauli proposed the famous Pauli exclusion principle, stating that each spatial orbital can accommodate a maximum of two electrons with opposite spin directions, characterized by m_s of $\frac{1}{2}$ or $-\frac{1}{2}$.⁶⁴ In electron-pair bonds, electrons can exist in a singlet state, where spins are antiparallel, the spin wavefunction is antisymmetric, and the spatial part of the wavefunction is symmetric. Alternatively, they can occupy a triplet state, where spins are parallel, the spin

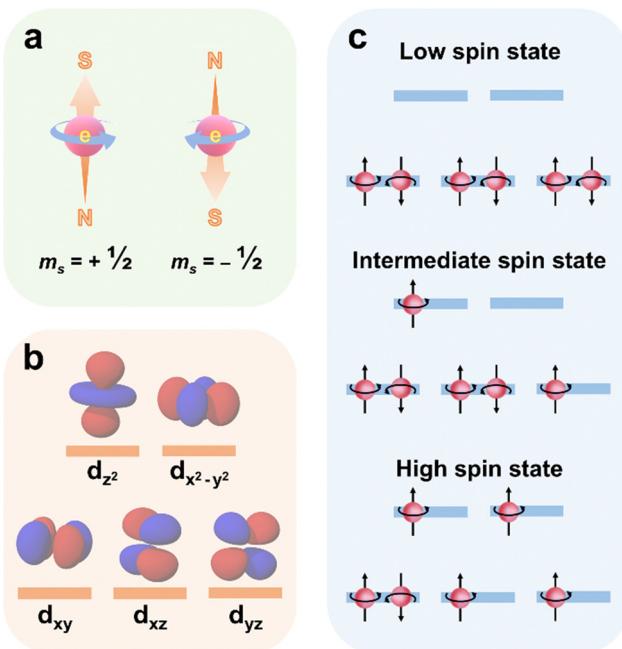


Fig. 4 Schematic diagram of electron spin and spin state. (a) Spin-up and spin-down electrons. (b) Spatial illustration of the five d-orbitals. (c) Crystal field splitting of d-orbitals in a metal cation.

wavefunction is symmetric, and the spatial part of the wavefunction is antisymmetric. This principle serves as the cornerstone for comprehending the electronic structure of atoms and the organization of elements in Mendeleev periodic table. At the end of 1925, George Uhlenbeck and Samuel Goudsmit introduced the concept of electron spin, which refers to the intrinsic angular momentum of an electron, independent of its orbital angular momentum.⁶⁵ After that, Friedrich Hund stated that electrons preferentially occupy different orbitals with parallel spins because of the minimum energy principle.⁶⁶ Hund's rules predict and explain the spin configuration of electrons, laying the foundation for the development of electron spin state. However, electron spin and spin state was not incorporated into quantum theory until 1928, when Paul A. M. Dirac published the Dirac equation to further explain the spin and magnetic moment of electrons.⁶⁷ In the decades that followed, research on electron spin and spin states spurred the advancement of spin-related fields, such as spintronics and quantum computing.

3.2 Electron spin states in photocatalysis

In recent years, the electron spin states in photocatalysis have garnered significant attention due to their notable impacts on the photocatalytic performance.^{12,13,21,68} Typically, the central metal atom in metallic photocatalysts forms an octahedral coordination structure with six surrounding ligand atoms. In this arrangement, the electrons in the metal's d orbitals repel the electrons of the ligands, causing those electrons closer to the ligands to have higher energy than those further away. As a result, the d orbitals split into two groups with distinct energy levels: the d_{xy} , d_{xz} , and d_{yz} orbitals, known as the t_{2g} orbitals,

have lower energy; while the d_{z^2} and $d_{x^2-y^2}$ orbitals, referred to as the e_g orbitals, have higher energy (Fig. 4b).

The splitting of metal d orbitals can alter the balance between crystal field splitting energy (the energy required to split the metal d orbitals) and spin pairing energy (the energy needed for two electrons to occupy the same orbital), leading to changes in the electron spin state. The spin states are categorized as low, intermediate, and high spin states (Fig. 4c). Generally, when the crystal field splitting energy exceeds the spin pairing energy, electrons first fill the t_{2g} orbitals with two electrons of opposite spins. Then, electrons occupy the e_g orbitals singly until the energy needed to fill an additional orbital surpasses the spin pairing energy. Conversely, if the crystal field splitting energy is lower than the spin pairing energy, electrons occupy all orbitals with similar energy levels singly before pairing up. As systems tend to minimize their total energy, those with crystal field splitting energy greater than spin pairing energy typically exhibit low spin configurations, while those with lower crystal field splitting energy favor high spin configurations.

3.3 Techniques for investigating electron spin state

The study of electron spin state is crucial for understanding spin-dependent chemistry. Commonly used techniques include X-ray absorption spectroscopy (XAS), Mössbauer spectroscopy, electron-energy-loss spectroscopy (EELS), electron spin resonance (ESR) spectroscopy, temperature-dependent magnetization (*M-T*) measurement, and theoretical modeling and calculations (Fig. 5).

3.3.1 X-Ray absorption spectroscopy. XAS analyzes the electronic structure and local chemical environment of materials.

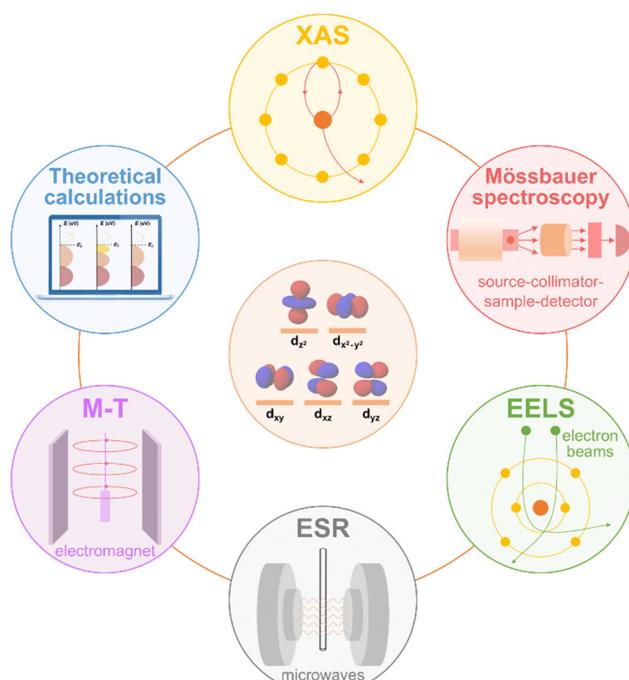


Fig. 5 Techniques for studying electron spin state.



When X-rays hit a sample, they can excite electrons to higher energy levels if their energy matches the binding energy. XAS consists of X-ray absorption near-edge structure spectroscopy (XANES) and extended X-ray absorption fine structure (EXAFS). The XANES detects 30–50 eV above the absorption edge to provide information on electronic state density and chemical state, while the EXAFS probes 50–1000 eV above the absorption edge to reveal local atomic structure. Moreover, the EXAFS can further reveal coordination numbers and bond lengths through Fourier transformation (FT) or wavelet transformation (WT) analysis. The XAS spectrum shape and normalized intensity of the metal L-edge peak provide insights into the spin state.⁶⁹ For example, a decrease in Co L-edge peak intensity indicated a spin state transition in Co^{III} from high spin to intermediate spin.¹⁵ Moreover, *in situ* XAS was performed to demonstrate that Co^{III} in Fe–CoOOH transitioned from low spin to intermediate spin.⁷⁰ Additionally, since the spin state affects the metal and O hybridization peaks in O K-edge, XAS can be used to quantify the metal–O bond covalency. For instance, LaCoO₃(111) with low spin Co^{III} showed a strong hybridization between the unoccupied e_g orbitals of Co^{III} and the O 2p orbitals, yet in LaCoO₃(100), the Co^{III} transitioned to intermediate spin and weakened the hybridization.⁷¹

In general, XAS offers high-resolution energy distribution that aid in determining the electron density of states and local spin state, particularly at the K-edge and L-edge absorption peaks. However, for metal centers with multiple coordination types, quantifying their coordination is challenging, as XAS provides average values for all absorbed atoms instead of a specific single site.¹⁴

3.3.2 Mössbauer spectroscopy. For Fe species, ⁵⁷Fe Mössbauer spectroscopy is an important technique to identify their spin configurations. When irradiated by γ -rays, the energy levels of Fe nucleus transition between the ground state and the first excited spin state, which are influenced by their electromagnetic and electrical environment and can be monitored by resonance fluorescence.¹⁴ Different spin states of Fe ions can be identified by their distinct isomer shift and quadrupole splitting values.⁷² Generally, a lower isomer shift indicates the low spin state of both Fe^{II} and Fe^{III}, while a lower quadrupole splitting value suggests the low spin state for Fe^{II} and the high spin state for Fe^{III}.¹⁵ For example, Mössbauer spectroscopy was applied to trace the low-to-intermediate spin transition of Fe^{III}, demonstrating the spin state transition mediated by axial Fe–O–Ti ligands.⁷³ Moreover, *in situ* ⁵⁷Fe Mössbauer spectroscopy was used to analyze the valence changes in high spin Fe^{II} and intermediate/low spin Fe^{III} during potentiostatic operation.⁷⁴ The results indicated that the Fe^{III} moiety irreversibly transitioned to Fe^{II}, while the Fe^{II} moiety remained stable for 50 h.

Mössbauer spectroscopy provides advantages including strong resistance to interference and applicability to various sample types such as powders, thin films, and bulk materials. However, it is limited to the analysis of ⁵⁷Fe and ¹¹⁹Sn and is primarily used to study the spin states in Fe-based catalysts.

3.3.3 Electron-energy-loss spectroscopy. In EELS, a known-energy electron beam is directed at a material, causing some

electrons to lose energy through inelastic scattering. By analyzing this energy loss, the electron spin configurations within the atoms can be determined. For instance, analyzing the peak intensity and energy of the EELS O K-edge spectra can differentiate between the spin states of Co^{III} in the bulk and surface phases of LaCoO₃.⁷⁵ A lower peak intensity around 530 eV indicated a higher spin state of Co^{III} in LaCoO₃. Additionally, the EELS Co L-edge spectra was used to determine the spin state transition of Co^{III}.⁷⁶ A decrease in Co valence and an increase in O vacancy concentration in Co–O layer was confirmed by the Co L₃ and Co L₂ peak intensities, indicating a spin state transition of Co^{III} from high spin to low spin.

EELS offers high energy resolution, enabling precise differentiation of spin states within the local band structure. When paired with transmission electron microscopy (TEM), it provides nanometer-scale spatial resolution for detailed spin state analysis in localized regions of a material. However, EELS requires samples to be sufficiently thin and stable.

3.3.4 Electron spin resonance spectroscopy. ESR is a powerful technique for studying the electron spin states in paramagnetic substances with unpaired electrons. When electrons in the sample resonate under a specific magnetic field and radiation frequency, changes in magnetic susceptibility produce an ESR signal. The ESR spectrum records the absorption intensity at different magnetic fields and radiation frequencies, revealing information about the spin state. However, measuring transition metal ions at room temperature can be problematic because some metal ions have short relaxation times, that is, the time for an excited spin system to return to the ground state.⁷⁷ This may result in broader spectra and exceed detection limits.¹⁴ Thus, lower test temperatures (below 77 K) are typically used for probing the spin states of metal-based materials. For example, low-temperature ESR spectra showed an increased peak intensity at $g = 2.3$ in Bi-doped SrCo_{3– δ} compared to unmodified SrCo_{3– δ} , indicating a higher proportion of intermediate spin and high spin Co^{III} in Bi-doped SrCo_{3– δ} .⁷⁸

ESR spectroscopy is highly sensitive to paramagnetic substances, including free radicals, transition metal ions, and certain metal oxides, even at extremely low concentrations. It generally does not damage the sample and can be used across various states (e.g., solids, liquids, and gases) and conditions (e.g., room and low temperatures). However, ESR is limited to samples with unpaired electrons and cannot provide information about systems with fully paired electrons.

3.3.5 Temperature-dependent magnetization measurement. M – T measurements assess the electron spin states of metal-based ferromagnetic materials in a controlled magnetic field under field-cooling procedures.¹⁵ Once the temperature exceeds the Curie temperature (T_C), the ferromagnetic materials lose their permanent magnetism and exhibit paramagnetic behavior. In the paramagnetic region, the susceptibility follows the Curie–Weiss law: $\chi = C/(T - T_C)$, where C is the Curie constant. Then, the effective magnetic moment μ_{eff} can be obtained by $\mu_{\text{eff}} = \sqrt{8C}\mu_{\text{B}}$, where μ_{B} is the Bohr magneton with a value of $9.274 \times 10^{-24} \text{ J T}^{-1}$. After getting μ_{eff} , the effective spin quantum number S_{eff} can be



calculated by $S_{\text{eff}} = \mu_{\text{eff}}/(g\mu_{\text{B}})$, where g is the Landé factor.^{14,15} S_{eff} determines the arrangement of electrons in the atomic orbits and the spin state transitions. For example, the spin states of Co^{III} were determined using M - T measurements at a 1000 Oe magnetic field under field-cooling procedures, demonstrating a spin state transition from low spin to high spin in Co^{III}.⁷⁹

M - T measurements reveal the magnetic phase transition energy of a material across different temperatures and identify temperature-dependent spin state transitions, such as the shift from paramagnetism to ferromagnetism. However, M - T measurements are primarily applicable to samples exhibiting paramagnetism or ferromagnetism and have limited use for nonmagnetic materials. Additionally, for samples with T_{C} exceeding 800 K, the Curie constant may not be readily accessible.

3.3.6 Theoretical modeling and calculations. Theoretical modeling and calculations are powerful techniques for studying spin-dependent chemistry. They can assist in identifying transitions between different spin states by comparing the total energies of the spin states. They also reveal the underlying reaction mechanisms by calculating and comparing the Gibbs free energy of reaction intermediates and products. For example, density functional theory (DFT) calculations revealed that the low spin Fe^{III} in NiFeOOH enhanced its catalytic activity. However, as the Fe^{III} content increased, a spin state transition from low spin to high spin occurred, which decreased the activity of surface Fe ions.⁸⁰ Moreover, DFT calculations were employed to investigate the origin of the high spin Co^{III} in CoOOH, indicating that this high spin state was associated with a decrease in Co valence state.⁸¹ The partial density of states (PDOS) results indicated that, after introducing high spin Co^{III} into CoOOH, the density of electronic states in the Co 3d and O 2p orbitals near the Fermi level increased significantly, promoting electron transfer during the catalytic process.

Theoretical modeling and calculations can predict the electronic structures and chemical properties of materials prior to

experiments, thereby guiding experimental design and optimizing materials. Additionally, they can forecast the chemical properties of materials under extreme conditions that may be challenging to achieve experimentally. However, the models used in theoretical calculations may not perfectly align with actual experimental conditions, potentially limiting the credibility of the results.

3.4 The role of electron spin control in photocatalysis

In recent years, electron spin control has driven notable advances in photocatalysis by: (i) enhancing light absorption of photocatalysts through tuning their energy band structures, enabling effective utilization of visible to NIR light; (ii) promoting charge separation by spin polarization effect, which accelerates the migration of photogenerated electrons and holes to the photocatalyst surface; and (iii) improving surface reaction kinetics through strengthening the interaction between photocatalysts and reactants and increasing the selectivity of target products (Fig. 6).

3.4.1 Enhancing light absorption. The energy band structure of a photocatalyst dictates its ability to absorb light. Specifically, a wider band gap means a larger energy difference between the CB and VB, requiring the absorption of higher-energy photons (short-wavelength light, such as UV) to excite electrons from the VB to the CB. As a result, photocatalysts with wide band gaps are only responsive to high-energy photons, limiting their absorption to a narrow portion of the solar spectrum and excluding much of the visible and NIR regions. Conversely, semiconductors with narrower band gaps require less energy for electron transitions and can be excited by low-energy photons (long-wavelength light, such as visible and NIR). Notably, the energy band structures of photocatalysts can be modified through electron spin control by inducing the spin-orbit coupling (SOC) effect.⁸² This effect arises from the interaction between an electron's spin and its orbital motion, which may lead to additional splitting of the CB and

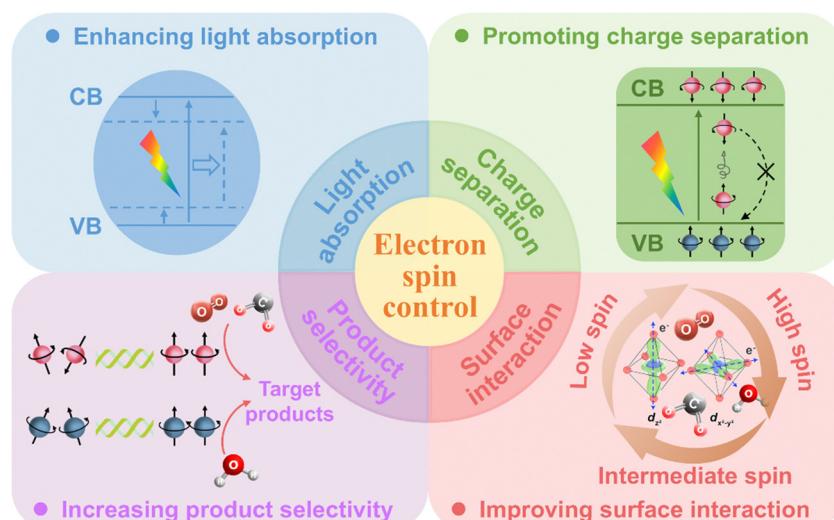


Fig. 6 The role of electron spin control in photocatalysis.



VB or the formation of additional sub-bandgap states. These changes rearrange the energy band structure and affect light absorption across different wavelengths. Additionally, the SOC effect can create new electron transition channels with distinct energy requirements, potentially enhancing light absorption within specific wavelength ranges.⁸³ For instance, spin-forbidden transitions can be induced by the SOC effect.¹⁸ Typically, photocatalysts undergo spin-allowed excitation, transitioning from the ground state to the excited singlet state and then to the excited triplet state ($S_0 \rightarrow S_1 \rightarrow T_1$) (Fig. 7a).¹⁸ However, this process involves intersystem crossing (ISC), which results in some energy loss and limits the effective excitation of electrons. The SOC effect in Os(II) polypyridine complexes enabled spin-forbidden excitation ($S_0 \rightarrow T_1$) under deep red and NIR light (660–800 nm) irradiation (Fig. 7b), significantly reducing ISC-related energy loss and extending the photocatalyst's absorption range from 615 nm to 730 nm (Fig. 7c). Moreover, the rearrangement of electron spin states in photocatalysts due to the SOC effect can lead to Zeeman splitting of energy levels, particularly when an external magnetic field is applied (Fig. 7d).¹⁹ This alters the energy required for electrons to transition from the ground state (VB) to the excited state (CB), thereby regulating the light absorption range.

Traditional methods to enhance the light absorption of photocatalysts typically involve: (i) reducing the band gap energy through metal and non-metal doping (e.g., N, S, Fe, and Co) to enable absorption of longer-wavelength light; (ii) introducing lattice defects, such as oxygen and metal vacancies, to create new energy levels or band states, thereby increasing the absorption efficiency of visible-to-NIR light; and (iii) modifying the photocatalyst surface with photosensitizers (e.g., dyes and quantum dots) to boost light absorption *via* energy or electron transfer mechanisms. In contrast, electron spin control offers a novel

approach by directly tuning the energy band structure through the SOC effect, without the need for complex material modifications. This presents a robust yet simple strategy for enhancing light absorption in visible and NIR regions and optimizing photocatalytic performance.

3.4.2 Promoting charge separation. Spin polarization effect, which biases electrons towards one of the spin states (spin up or spin down) rather than an equal distribution between both states, can reduce the recombination of photo-generated electrons and holes.^{12,13,84} Specifically, when a spin-up electron is excited to the CB, the hole left in the VB retains the same spin orientation. However, during the electron transfer process, factors such as the SOC effect, hyperfine interactions, and external magnetic fields may cause the electron's spin to flip (e.g., from spin up to spin down), while the hole's spin remains unchanged. Because electrons and holes with opposite spin directions cannot recombine directly *via* photoluminescence or heat dissipation, this spin-forbidden effect significantly lowers the probability of recombination.^{13,20,84} Consequently, in a highly spin-polarized environment, electron-hole recombination is effectively suppressed, enhancing the charge separation efficiency of a photocatalyst. For example, the photocatalytic CO_2 reduction performance of CsPbBr_3 was improved by 5.7 times due to the promoted charge separation, which was achieved by creating a spin-polarized environment through Mn doping and the application of an external magnetic field (Fig. 8a).²⁰ Pristine CsPbBr_3 did not exhibit an extended carrier lifetime under a magnetic field (Fig. 8b). Interestingly, Mn-doped CsPbBr_3 ($\text{Mn}-\text{CsPbBr}_3$) generated spin-polarized electrons, and the electron spin polarization in $\text{Mn}-\text{CsPbBr}_3$ was further enhanced under a magnetic field, resulting in an extended lifetime of photogenerated carriers (Fig. 8c).

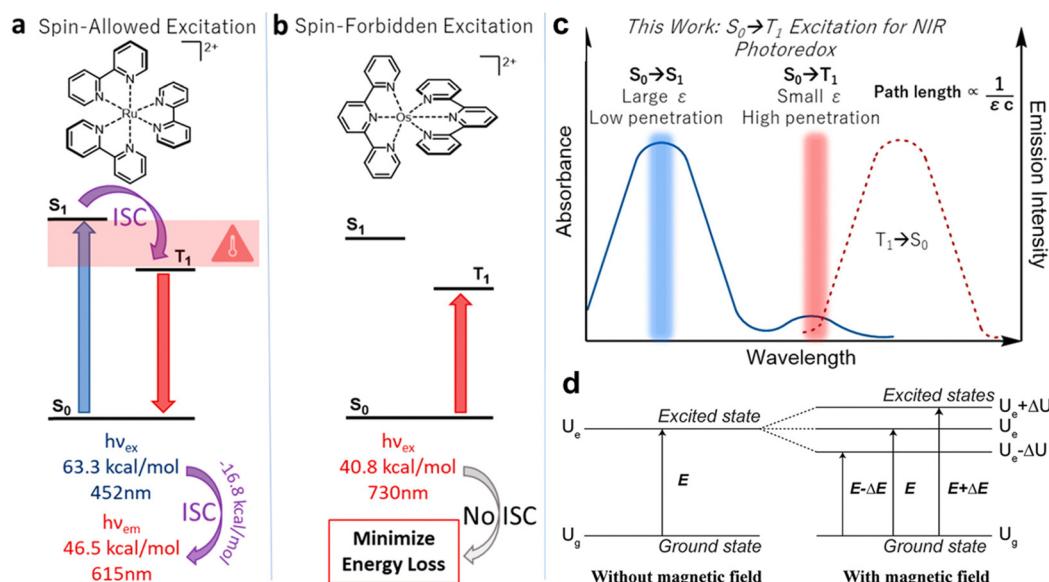


Fig. 7 Enhanced light absorption by the regulation of energy band structure. Schematic illustration of (a) spin-allowed excitation and (b) spin-forbidden excitation. (c) Comparison of light penetration in spin-allowed excitation and spin-forbidden excitation. Reproduced with permission from ref. 18. Copyright 2020 American Chemical Society. (d) Energy splitting introduced by Zeeman effect. Reproduced with permission from ref. 19. Copyright 2022 Cell Press.



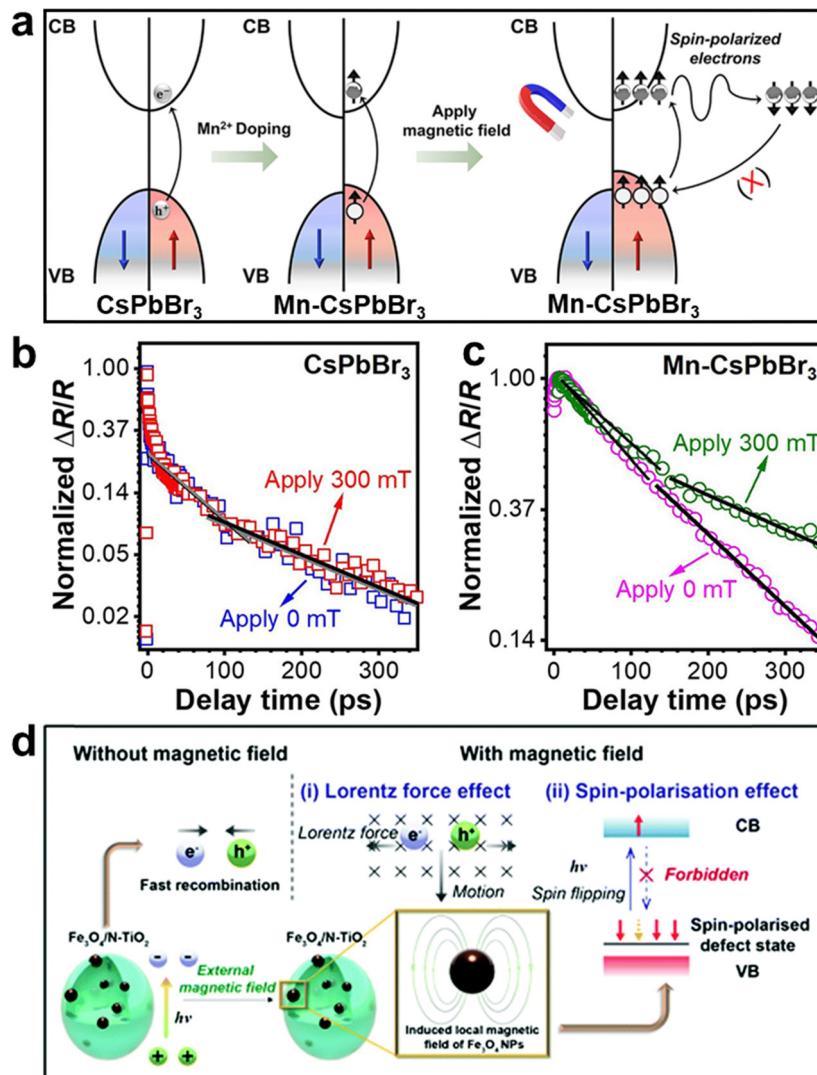


Fig. 8 Promoted charge separation by the application of an external magnetic field. (a) Schematic illustration of the spin polarization induced-longer photoexcited carrier lifetime in Mn–CsPbBr₃ under an external magnetic field. The normalized photoinduced transient reflectivity changes ($\Delta R/R$) of (b) CsPbBr₃ and (c) Mn–CsPbBr₃ with and without an external magnetic field. Reproduced with permission from ref. 20. Copyright 2022 American Chemical Society. (d) Schematic illustration of the magnetic field promoted photocatalysis system. Reproduced with permission from ref. 21. Copyright 2022 Royal Society of Chemistry.

Thus, the combination of Mn doping and an external magnetic field effectively increased the number of spin-polarized carriers, suppressed charge recombination, and enhanced photocatalytic activity.

Another study revealed that Au-loaded Fe₃O₄/N-TiO₂ superparamagnetic photocatalyst significantly enhanced the efficiency of photocatalytic water splitting under an external magnetic field because of the promoted charge separation.²¹ This promotion was attributed to the combined effects of the Lorentz force and spin polarization effect (Fig. 8d). In this system, the local magnetic field generated by Fe₃O₄ nanoparticles aligns the magnetic moments in N-TiO₂ in parallel, creating a highly spin-polarized environment. Under an external magnetic field, electrons excited to the CB undergo spin relaxation, changing from their initial spin state (spin down) to

an alternative state (spin up). Due to the scarcity of spin-up holes, these spin-up electrons encounter more difficulty returning to the VB, thereby reducing charge recombination and boosting photocatalytic performance.

In addition to the application of external magnetic fields, creating material defects can also regulate electron spin, thereby promoting charge separation.^{13,23} For example, cationic defect-rich Bi₄Ti₃O₁₂ induced spin polarization, resulting in the generation of spin-parallel photogenerated electrons, which inhibited electron–hole recombination and led to efficient O₂ activation (Fig. 9a).²³ The reduced photoluminescence (PL) intensity in steady-state PL spectra suggest that the defects in Bi₄Ti₃O₁₂ suppressed charge recombination (Fig. 9b). Meanwhile, time-resolved photoluminescence (TRPL) decay spectra reveal that defect-rich Bi₄Ti₃O₁₂ facilitated a higher

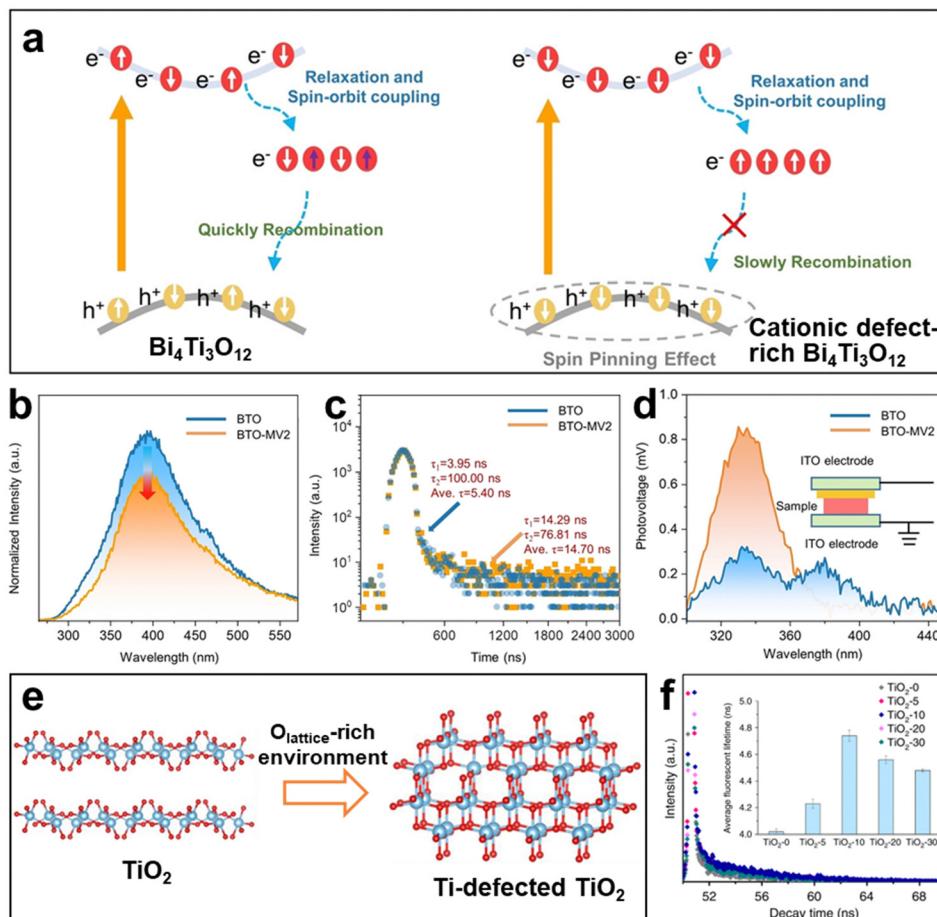


Fig. 9 Promoted charge separation by defect engineering. (a) Schematic illustration of the excited carrier dynamics in $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ and cationic defect-rich $\text{Bi}_4\text{Ti}_3\text{O}_{12}$. (b) Steady-state PL spectra, (c) TRPL decay spectra, and (d) SPV spectra of $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ and cationic defect-rich $\text{Bi}_4\text{Ti}_3\text{O}_{12}$. Reproduced with permission from ref. 23. Copyright 2023 John Wiley & Sons. (e) Schematic illustration of the synthesis method of Ti-defected TiO_2 . (f) TRPL decay spectra of TiO_2 and Ti-defected TiO_2 . Reproduced with permission from ref. 13. Copyright 2020 Springer Nature.

participation of photogenerated electrons and holes in the reaction, confirming the more effective charge separation (Fig. 9c). Moreover, steady-state surface photovoltage (SPV) spectroscopy indicates that the local electric field intensity of defect-rich $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ was approximately three times higher than that of $\text{Bi}_4\text{Ti}_3\text{O}_{12}$, demonstrating the improved charge separation in the defect-rich $\text{Bi}_4\text{Ti}_3\text{O}_{12}$ (Fig. 9d). In another study, intrinsic Ti vacancies in TiO_2 were created by an oxygen-rich environment during the thermal assembly of Ti–O–Ti parallel lattice chains (Fig. 9e).¹³ These Ti vacancies resulted in the parallel alignment of electron spin orientations. Interestingly, varying the concentration of Ti vacancies allowed for tuning the degree of spin polarization. Compared to the pristine TiO_2 , Ti-vacancy-modified samples showed extended average TRPL lifetimes (Fig. 9f), suggesting the promoted charge separation by the spin polarization effect.

Traditional methods for promoting the separation of photogenerated charge carriers generally focus on manipulating the structures, surface properties, and heterojunctions of semiconductors. Common strategies include: (i) constructing heterojunctions between different semiconductors (e.g., p–n junctions,

type-II and Z-scheme heterojunctions) to utilize energy band alignment for efficient separation of photogenerated electrons and holes; (ii) depositing noble metal nanoparticles (e.g., Pt, Au, and Ag) and conductive carbon materials (e.g., graphene, carbon dots, and carbon nanotubes) on photocatalyst surface to form “electron capture” layers that enhance electron separation efficiency; (iii) optimizing charge separation efficiency by controlling the exposure of specific crystal facets of photocatalysts; and (iv) introducing lattice defects to create “electron traps”, which facilitate charge separation. In contrast to these approaches, electron spin control reduces the recombination probability of electrons and holes through the spin polarization effect, thereby extending carrier lifetimes and enhancing photocatalytic performance. This method controls charge separation by adjusting the spin state and spin-related interactions within photocatalysts, simplifying the design and fabrication process while offering a more precise and controllable way to enhance photocatalytic efficiency.

3.4.3 Strengthening the interaction between photocatalysts and reactants. The surface reaction steps involve the adsorption of reactants, charge transfer between the photocatalysts and

reactants, and the desorption of reaction products. Among these steps, the adsorption of reactants by photocatalysts directly impacts the efficiency of photocatalytic reactions. Moreover, the active centers of photocatalytic reactions, which are often localized in specific regions such as metal active centers, defect sites, and acidic/basic sites, play a vital role in determining the efficiency of surface reaction. Notably, the adsorption capacity of photocatalysts is closely linked to the spin states of active centers.¹⁰ Adsorption energy represents the interaction between active centers and reactants or intermediates adsorbed on it. More negative adsorption energy indicates a more stable interaction between them, which can reduce the reaction energy barrier and facilitating the reaction rate. Generally, adsorption energy is influenced by the orbital hybridization between the adsorbate and catalyst, which can be modulated by adjusting the spin states of the active centers. According to the d-band center theory, when an adsorbate approaches the metal surface, its atomic orbitals interact with the metal's d orbitals, leading to energy level splitting.⁸⁵ The bonding orbitals formed are lower in energy and more stable, while the antibonding orbitals are higher in energy and less stable. If electrons predominantly occupy the bonding orbitals, the system's overall energy decreases, strengthening the bonding. Conversely, if more electrons fill the antibonding orbitals, the structure becomes less stable, and the bonding strength weakens. Adsorption energy can be assessed by the position of the d-band center and the

degree of antibonding orbital occupancy.⁸⁵ When the metal's d-band center is sufficiently high, the antibonding orbital will be above the Fermi level and remain unoccupied due to insufficient electron flow, which facilitates adsorption. However, a low d-band center can position the antibonding orbital below the Fermi level, leading to its occupation by incoming electrons and thereby hindering adsorption.

Modulating the electron spin states of metal centers enables precise control over the adsorption energy, thereby improving the surface reaction kinetics.¹⁴ For example, the spin state of Co in the covalent organic framework (COF)-367-Co can be controlled by adjusting the oxidation state of Co at the porphyrin center.⁶⁸ This alteration affected the electron distribution and the orientation of the Co-3d orbitals, thereby influencing the interaction between Co and CO₂ molecules. COF-367-Co^{III}, compared to COF-367-Co^{II}, exhibited superior photocatalytic CO₂ reduction activity and markedly enhanced selectivity for formic acid (HCOOH) (Fig. 10a). DFT calculations reveal that in COF-367-Co^{II}, Co-3d_{xz} or Co-3d_{yz} orbitals couple with the O-2p orbitals of CO₂, whereas in COF-367-Co^{III}, Co-3d_{z²} orbitals interact with the O-2p orbitals, resulting in the adsorption energy on COF-367-Co^{III} more than double that of COF-367-Co^{II} (Fig. 10b). Moreover, COF-367-Co^{III} exhibits a stronger interaction with HCOOH than COF-367-Co^{II}, favoring the formation of HCOOH while inhibiting its further conversion, thereby increasing the selectivity for HCOOH. In another study,

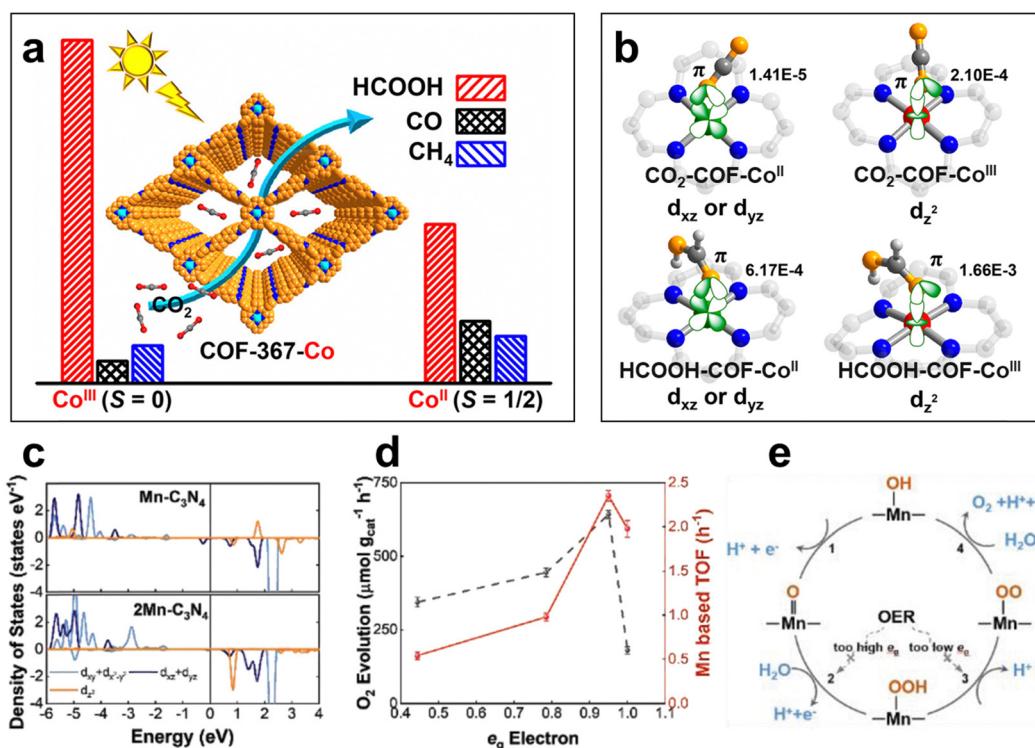


Fig. 10 Strengthened interaction between photocatalysts and reactants by the regulation of metal spin state. (a) Photocatalytic CO₂ reduction activities by COF-367-Co with different Co spin states. (b) Different coupling modes of CO₂ and HCOOH interacting with Co site at different spin states. Reproduced with permission from ref. 68. Copyright 2020 American Chemical Society. (c) Partial density of states of Mn obtained by DFT calculation. (d) Correlation of e_g occupancy and oxygen evolution values. (e) Proposed four-electron mechanism for oxygen evolution over Mn sites. Reproduced with permission from ref. 24. Copyright 2019 John Wiley & Sons.



Co^{II} with low spin state has an empty e_g orbital, making it difficult to deprotonate oxyhydroxides to form peroxide ions, which severely inhibited the reaction efficiency of oxygen evolution.^{86,87} However, Co^{III} with intermediate spin state achieved optimal adsorption of intermediates (e.g., *OOH, *O, and *OH), resulting in excellent activity. Also, tuning the spin state of the Fe center can alter its adsorption energy for reactants. For instance, when Fe^{III} transitioned from low spin state to high spin state, the increased electrons in the e_g orbitals weakened the adsorption of O₂, leading to poor oxygen reduction performance.⁸⁸ Conversely, when transitioned to intermediate spin state, the d_{2z} orbitals of Fe^{III} easily bound to the antibonding π orbital of O₂, thereby exhibiting enhanced oxygen reduction efficiency.

Additionally, regulating the spin state of the Mn active center in Mn–C₃N₄ can modulate the interaction between the Mn–C₃N₄ and the reaction intermediates (e.g., *OOH, *O, and *OH), thereby enhancing the photocatalytic oxygen evolution performance.²⁴ DFT calculations indicate that with a single Mn atom, Mn is in a high-spin state, whereas with two Mn atoms, the spin state of Mn is reduced (Fig. 10c). Consequently, the spin state of Mn can be tuned by adjusting the Mn concentration in Mn–C₃N₄. The binding strength between Mn and the intermediates is influenced by the antibonding e_g occupancy, with the optimal e_g occupancy around 0.95 (Fig. 10d). For photocatalysts with high e_g occupancy, the Mn–O bond is too weak, which impedes the formation of the O–O bond. In contrast, the Mn–O bond of photocatalysts with low e_g occupancy is too strong, hindering proton removal from Mn–OOH (Fig. 9e). Therefore, by precisely adjusting the spin state of Mn active center to achieve moderate e_g filling, the balance between these competing processes can be optimized to enhance photocatalytic activity.

The interaction between photocatalysts and reactants is also crucial in determining the efficiency of charge transfer between photocatalysts and reactants. When reactant molecules coordinate with metal catalysts, the metal center's spin state can influence the energy distribution of the reactants' highest occupied molecular orbital (HOMO) and lowest unoccupied molecular orbital (LUMO).⁸⁹ The HOMO represents the molecule's ability to donate electrons, while the LUMO reflects its ability to accept electrons. Generally, a smaller difference between the HOMO and LUMO energy levels facilitates the reaction. Consequently, regulating the surface interaction can directly affect the electron transfer between reactants and catalysts, thereby accelerating or slowing down the reaction rate. For instance, improving the adsorption of surface reactants (e.g., O₂ and peroxide) on catalysts can regulate the electronic structures of these reactants, thereby enhancing electron transfer between the reactants and the metal center.⁹⁰ Moreover, in organic synthesis, adjusting the adsorption capability of metal catalysts enables the design of reactants with optimized HOMO and LUMO energy levels, which lowers the reaction energy barrier.⁹¹

3.4.4 Increasing the selectivity of target products. The products of chemical reactions are often closely related to the electron spin states of reactants and intermediates, allowing for the selectivity of desired products *via* electron spin control. Generally, reactions that produce triplet O₂ from singlet OH[–]

are thermodynamically forbidden, which is the rate-limiting step of oxygen evolution.¹² Recently, the spin polarization effect has demonstrated potential for the selective generation of O₂ from OH[–].^{12,13,27} The spin polarization effect induced by photocatalysts unified the spin direction of HO[•] intermediates, which reduced the formation of singlet byproduct H₂O₂ and increased the selectivity for triplet O₂. For example, chiral ZnO synthesized with chiral methionine molecules as symmetry breakers triggered a chiral-induced spin selectivity (CISS) effect and enhanced photocatalytic oxygen evolution.¹² The chiral structure of ZnO served as a spin filter, leading to spin polarization of the photogenerated charge carriers, which not only extended the carrier lifetimes but also improved the selectivity for O₂ during the reaction. Compared to achiral ZnO, chiral ZnO exhibited a higher electron transfer number in the oxygen evolution reaction, indicating a more effective suppression of H₂O₂ formation (Fig. 11a). During photocatalysis, a distinct H₂O₂ peak was detected in achiral ZnO, whereas chiral ZnO showed a significantly reduced H₂O₂ peak, confirming a substantial reduction in H₂O₂ formation (Fig. 11b). Furthermore, EPR spectra reveal that the HO[•] peak intensity in chiral ZnO was approximately twice that of achiral ZnO, indicating that polarized HO[•] was more stable and less likely to recombine to form H₂O₂ (Fig. 11c). These results demonstrate that polarized holes suppressed H₂O₂ generation by controlling the spin state of the HO[•] intermediate, ultimately enhancing the efficiency of oxygen evolution (Fig. 11d). In addition, manipulating the spin arrangement of electrons and holes allows for the selective stabilization of specific intermediates.⁹² For instance, aligning the electron spin of HO[•] in the same direction prevented them from interacting with each other, which extended their lifetime and enhanced the efficiency of organic pollutant degradation.^{12,13}

Currently, there are limited approaches for enhancing the surface reaction kinetics in photocatalysis, and these primarily focus on optimizing surface properties, active sites, and the adsorption of reactants. Common strategies include: (i) introducing additional active sites on photocatalyst surface (e.g., metal nanoparticles, acidic/basic sites, and oxygen/metal vacancies) to improve reactant adsorption; (ii) loading co-catalysts (e.g., Ni(OH)₂ and RuO₂) onto photocatalyst surface, which act as electron or hole capture centers to lower reaction energy barriers; and (iii) modifying surface functional groups (e.g., carboxyl, amino, and hydroxyl) to adjust surface properties, thereby optimizing the adsorption and desorption behaviors of reactants. Compared with these strategies, electron spin control offers a robust yet simple approach to not only improve the adsorption energy but also increase the product selectivity during photocatalytic processes by optimizing the electron spin states of active sites on photocatalyst surface.

4. Strategies for optimizing photocatalysis *via* electron spin control

In photocatalysis, the electron spin control strategy focuses on precisely manipulating the spin and spin state of electrons to



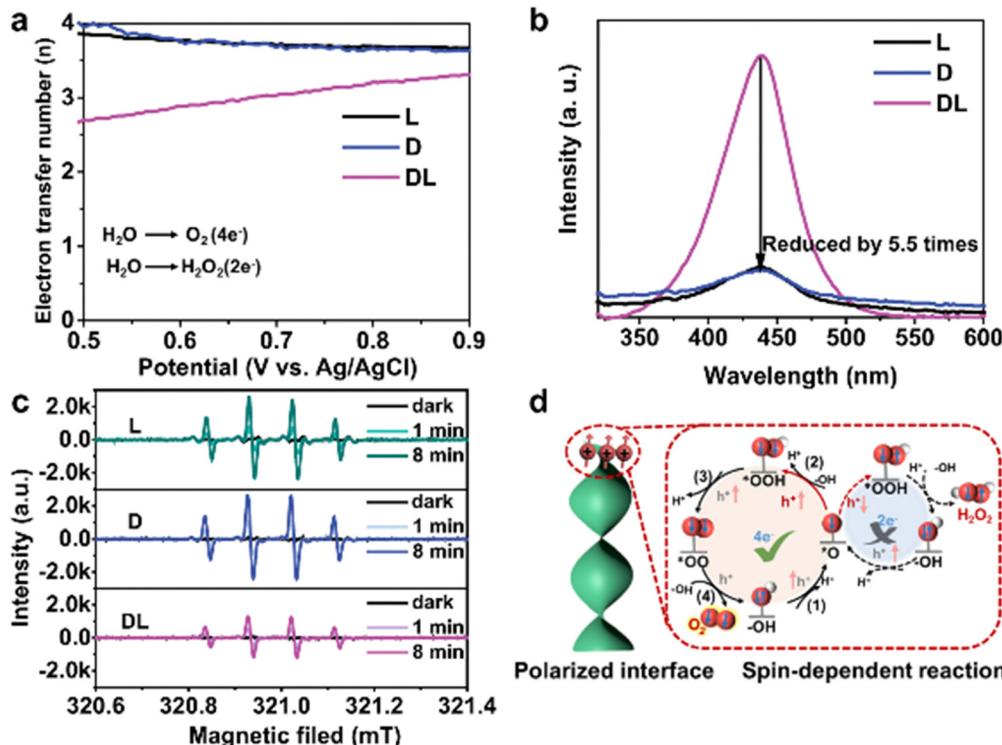


Fig. 11 Increased O_2 selectivity by CISS effect. (a) The calculated electron transfer number as a function of applied potentials via photoelectrochemical rotating ring-disk electrode measurement. (b) The amount of hydrogen peroxide with o -tolidine indicator detected by UV-vis absorption spectra. (c) EPR spectra of DMPO- HO^\bullet adducts in photocatalysis. (d) Schematic representation of spin-polarized surface reaction. Reproduced with permission from ref. 12. Copyright 2023 Springer Nature.

optimize the separation of photogenerated electron-hole pairs and their subsequent reactions on the photocatalyst surface, which can significantly enhance photocatalytic activity. These strategies typically involve doping design, defect engineering, magnetic field regulation, metal coordination modulation, chiral induced spin selectivity, and combined strategies (the combination of these strategies). Table 1 summarizes recent advancements in electron spin control strategies for various photocatalytic applications (*e.g.*, water splitting, CO_2 reduction, pollutant degradation, and N_2 fixation). We will provide specific examples illustrating the impact of electron spin control on photocatalytic light absorption (evaluated by band gap alteration), charge separation (characterized by TRPL average lifetime), and overall improvements in photocatalytic efficiency across various photocatalysts, along with detailed discussions of the underlying mechanisms.

4.1 Doping design

Doping design is a widely used modification technique for photocatalysts. It introduces metal and non-metal elements to alter the electronic energy levels and band structure of the photocatalyst. This can broaden the light absorption range and improve carrier separation efficiency, ultimately leading to increased photocatalytic performance.² In recent years, doping design has been employed to manipulate electron spin in photocatalysts. In certain metals and non-metals, unpaired electrons carry spin angular momentum. When elements with

unpaired electrons are doped into a photocatalyst, they create local magnetic moments within the material. These local magnetic moments can interact with the surrounding electron spins, influencing the overall electron arrangement and potentially leading to an ordered spin arrangement.^{93,100,135} This, in turn, can affect the electron transport properties during photocatalytic reactions. Furthermore, doping elements into the photocatalyst lattice can introduce new electronic energy levels or alter the existing energy level structure,¹³⁶ which may cause a rearrangement of electrons and result in the splitting of energy levels. The split energy levels will be preferentially occupied by electrons with different spin states, leading to a spin selection effect. As lower-energy spin states are more readily occupied by electrons, the energy level splitting induced by doping can cause electrons in the system to preferentially adopt specific spin states, resulting in overall spin polarization.¹³⁷

Doping magnetic elements (*e.g.*, Fe, Co, and Ni) into photocatalysts is an effective strategy for manipulating electron spin state. For example, Li *et al.* employed Fe-doped $\text{Bi}_4\text{O}_5\text{Br}_2$ (Fe- $\text{Bi}_4\text{O}_5\text{Br}_2$) to photoactivate H_2O_2 for the degradation of organic pollutants (*e.g.*, ciprofloxacin (CIP) and Cu-ethylenediaminetetraacetic acid (Cu-EDTA)) in water.⁹³ They optimized the spin interactions between Fe sites and coordinated O atoms (derived from H_2O_2) and increased high-spin Fe sites to generate $\text{O}_2^\bullet^-$ with high pH tolerance, thus enhancing Fenton-like degradation under neutral or alkaline environment (Fig. 12a). It was observed that pH had minimal impact $\text{O}_2^\bullet^-$ production



Table 1 Summary of strategies for electron spin control

Photocatalyst	Strategy	Photocatalytic application	Bandgap adjustment (eV)	TRPL lifetime regulation (ns)	Efficiency improvement	Ref.
Fe–Bi ₄ O ₅ Br ₂	Fe doping	Pollutant degradation	—	—	60 → 92% (25 min)	93
Fe–CN	Fe doping	Aniline hydrogenation	—	—	7.6 → 95.0% PhNH ₂	94
Co–BiVO ₄	Co doping	CO ₂ reduction	2.40 → 1.93	—	1.6 → 109.2 μmol per g per h CO	95
Co–BiOCl	Co doping	CO ₂ reduction	3.32 → 3.26	3.6 → 6.8	—	96
Cu–LaFeO ₃	Cu doping	Pollutant degradation	2.26 → 2.25	—	0.5 → 4.7 s ^{−1}	97
Cu–TiO ₂	Cu doping	Water splitting	3.20 → 3.06	—	—	98
Ti–CTF	Ti doping	Pollutant degradation	2.80 → 2.68	—	33.5 → 97.8 (100 min)	99
Mn–Co ₃ O ₄	Mn doping	CO ₂ reduction	—	—	0.8 → 23.4 μmol per g per h CH ₄	100
Mn–Fe ₂ O ₃	Mn doping	Pollutant degradation	—	—	78 → 92% (2 h)	101
Mn–CN	Mn doping	Water splitting	—	0.6 → 1.9	60.0 → 695.1 μmol per g per h H ₂	24
Mn–CN	Mn doping	Water splitting	2.79 → 2.55	17.5 → 4.1	0.4 → 13.5 mmol per g per h H ₂	102
Ni–BiOBr	Ni doping	CO ₂ reduction	—	2.6 → 3.6	86.8 → 378.7 μmol per g per h CO	103
O–ZnIn ₂ S ₄ /Ni ₁₂ P ₅	O doping	Water splitting	—	1.0 → 2.6	1.9 → 15.8 mmol per g per h H ₂	104
S–CTF	S doping	Pollutant degradation	—	—	9 → 100% (1 h)	105
B–CN	B doping	H ₂ O ₂ production	2.61 → 2.48	—	59 → 1710 μM h ^{−1}	106
I–GO	I doping	Water splitting	—	1.2 → 2.2	38.6 → 96.2 μmol per g per h H ₂	107
Se–CN	Se doping	Water splitting	3.06 → 2.73	20.0 → 16.3	0.2 → 5.4 μmol per g per h H ₂	108
P/N–CN	P and N co-doping	Water splitting	—	21.6 → 1.8	0.1 → 22.2 mmol per g per h H ₂	109
C/B–TiO ₂	C and B co-doping	—	2.90 → 2.10	—	—	110
Bi ₄ Ti ₃ O ₁₂	Ti defect	Pollutant degradation	—	5.4 → 14.7	—	23
TiO ₂	Ti defect	Water splitting	—	4.1 → 4.8	1.8 → 18.5 mmol per g per h H ₂	13
ZnO	Zn defect	Water splitting	—	2.7 → 3.7	0.1 → 5.8 mmol per g per h H ₂	111
WO ₃	O defect	U(vi) reduction	2.67 → 2.54	99.1 → 78.4	20.5 → 287.1 mg g ^{−1}	112
CdS	S defect	Water splitting	2.39 → 2.13	6.4 → 17.3	0.8 → 41.7 mmol per g per h H ₂	113
SnS ₂	S defect	Water disinfection	2.29 → 2.20	0.1 → 1.7	—	114
Zinc porphyrin	N defect	CO ₂ reduction	1.83 → 1.80	—	4.6 → 12.5 μmol per g per h CO	115
CdS	Cd and S defects	Water splitting	—	0.86 → 0.95	1.2 → 10.6 mmol per g per h H ₂	116
Ti-based MOF	Ti and organic linker defects	Water splitting	—	—	1.4 → 3746.7 μmol per g per h H ₂	117
Fe ₂ O ₃ /rGO	Magnetic field	Pollutant degradation	—	—	59 → 84% (40 min)	118
Co–MOF	Coordination regulation	CO ₂ reduction	—	—	—	119
Fe(Pz)[Pt ^{II/IV} –(CN) ₄ I _{1.2} ·2H ₂ O	Coordination regulation	H ₂ O ₂ production	2.35 → 1.86	—	0 → 66 mM g ^{−1} h ^{−1}	120
Au ₃ Fe ₁ /Mo	Coordination regulation	N ₂ fixation	—	—	60 → 484 μmol per g per h NH ₃	49
Chiral ZnO	CISS effect	Water splitting	—	1.8 → 4.2	0.8 → 1.3 mmol per g per h H ₂	12
Fe–CsPbBr ₃	Fe doping and magnetic field	CO ₂ reduction	—	13.3 → 2.9	19.8 → 33.3 μmol per g per h CO	121
Fe–BiVO ₄	Fe doping and magnetic field	Water splitting	—	3.9 → 5.5	0 → 61.5 μmol per g per h H ₂	122
Co–CsPbBr ₃	Co doping and magnetic field	Water splitting	2.54 → 2.13	—	581.7 → 1042.1 μmol per g per h H ₂	123
Co–CN	Co doping and magnetic field	Water splitting	—	—	12 → 3979 μmol per g per h H ₂	124
Mn–CsPbBr ₃	Mn doping and magnetic field	CO ₂ reduction	—	—	0.6 → 17.6 μmol per g per h CO	20
Ni–CdS/MoS ₂	Ni doping and magnetic field	Water splitting	—	5.9 → 11.0	339 → 1322 μmol per g per h H ₂	125
Co ₃ O ₄	Co defect and magnetic field	CO ₂ reduction	—	0.14 → 3.51	9.4 → 70.8 μmol per g per h CO	126
Bi ₂ WO ₆ /CuS	W, Cu defects and magnetic field	Pollutant degradation	—	3.27 → 3.60	74.5 → 90.5%	127
BaTiO ₃	O defect and magnetic field	N ₂ fixation	—	—	0.1 → 1.9 mg L ^{−1} h ^{−1}	50
CuInP ₂ S ₆	S defect and magnetic field	CO ₂ reduction	—	—	4.5 → 12.5 μmol per g per h CH ₄	128
Ni–CdS	Ni doping and S defect	Water splitting	2.37 → 2.26	—	3.6 → 20.3 mmol per g per h H ₂	129
K–CN/AgCl/ferrrihydrite	K doping and Ag regulation	Pollutant degradation	2.87 → 2.44	—	75 → 100% (30 min)	130
MoS ₂ /graphene	Heterostructure construction	Water splitting	—	6.0 → 27.7	0 → 240 μmol per g per h H ₂	131
CN/graphene	Heterostructure construction	Water splitting	2.51 → 2.42	6.8 → 4.7	0.9 → 14.2 μmol per g per h H ₂	132
Co–COF	Metal valence regulation	CO ₂ reduction	1.10 → 1.03	—	48.6 → 93.0 μmol per g per h HCOOH	68
Fe ₂ O ₃ /CN	Crystal size regulation	Pollutant degradation	2.75 → 2.47	4.2 → 3.8	0.04 → 0.21 min ^{−1}	133
FeS ₂ /CuCo ₂ O ₄	Pyroelectric field	Water splitting	—	5.7 → 15.0	1.9 → 19.5 mmol per g per h H ₂	134

(Fig. 12b), and Fe–Bi₄O₅Br₂ exhibited longer carrier lifetimes and superior CIP degradation compared to Bi₄O₅Br₂ (Fig. 12c). This is because the O–O bonds of H₂O₂ adsorbed on Fe–Bi₄O₅Br₂ showed increased resistance to scission, facilitating the generation of *OOH in the subsequent reaction and thereby enhancing the formation of O₂^{•−} (Fig. 12d). This work demonstrates that doping with high-spin Fe can enhance charge separation and promote the formation of surface

*OOH intermediates, ultimately boosting the generation of O₂^{•−} for more efficient pollutant degradation.

Another study introduced spin-polarized electrons into BiOCl nanosheets by doping with magnetic Co, which significantly improved the efficiency of photocatalytic CO₂ reduction and achieved a high hydrocarbon selectivity of 76.9% for converting CO₂ into CH₄ and C₂₊ products.⁹⁶ Compared to BiOCl, Co–BiOCl exhibited a narrower band gap (3.32 → 3.26 eV) and



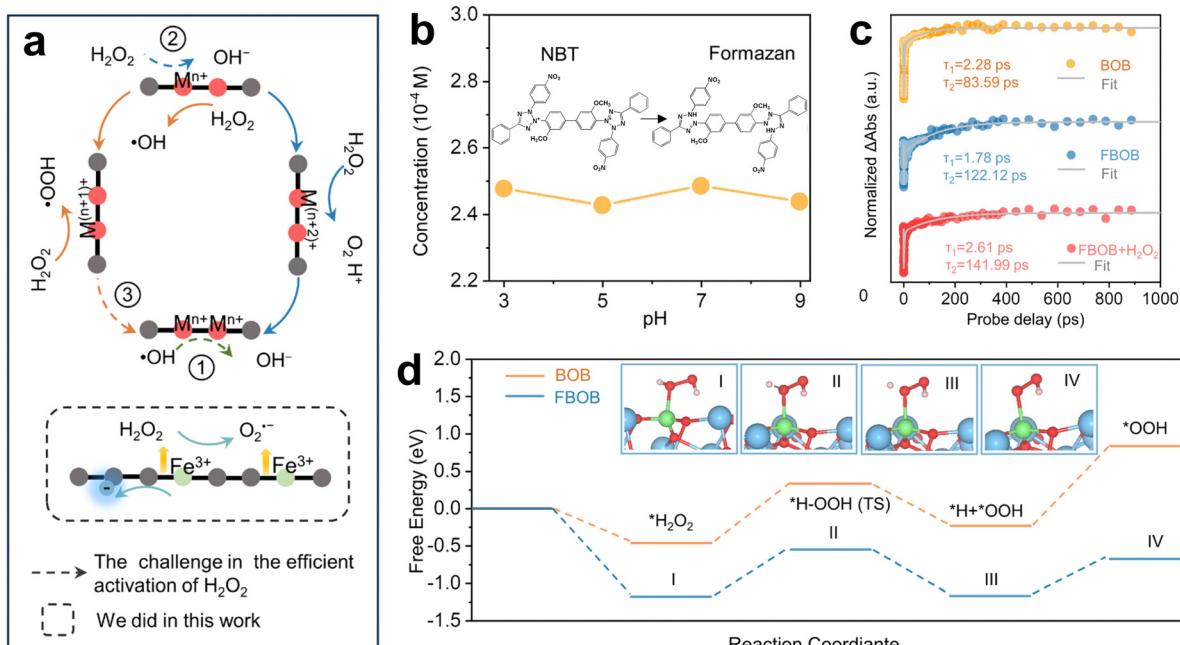


Fig. 12 Generation of $\text{O}_2^{\bullet-}$ mediated by high-spin Fe doping. (a) Schematic of the challenge in H_2O_2 activation and this work, M represents surface transition-metal atoms. (b) The production of $\text{O}_2^{\bullet-}$ at various pH levels. (c) Transient absorption spectroscopy (TAS) kinetics probed at 450 nm. (d) Reaction path of H_2O_2 activation and free energy profile of $\text{O}_2^{\bullet-}$ production process. Reproduced with permission from ref. 93. Copyright 2024 National Academy of Sciences.

longer carrier lifetime (3.6 → 6.8 ns). Mechanistic analysis reveals that the spin-polarized electrons enhanced CO adsorption and reduced the kinetic barrier for $^*\text{CH}_2$ formation, ultimately leading to highly selective hydrocarbon generation.

Single-atom doping can not only alter the electron spin environment of photocatalysts by introducing unpaired electrons but also provide a strategy to maximize atomic utilization, offering an efficient and sustainable strategy for optimizing photocatalysis. For instance, Zhu *et al.* developed a single-atom Ti-doped covalent organic framework (Ti-CTF) for the photocatalytic degradation of 2,2,4,4'-tetrahydroxybenzophenone (BP-2), with $\text{O}_2^{\bullet-}$ being the primary reactive species.⁹⁹ Compared to pristine CTF, the spin-polarized Ti-CTF significantly enhanced the adsorption and degradation of BP-2, achieving a degradation rate 17 times higher than that of CTF. Experimental results and theoretical calculations indicate that single-atom Ti, bound to pyridine and triazine N, induced electron spin-down polarization near the Fermi level, which promoted electron transfer and subsequent surface reactions (Fig. 13a). Moreover, Ti doping reduced the bandgap from 2.80 to 2.68 eV, thereby improving light absorption. This demonstrates the synergistic effect of electron spin control and Ti doping in optimizing photocatalytic performance. In another study, single-atom Mn was incorporated into Co_3O_4 photocatalyst to replace the octahedral Co (Fig. 13b), inducing electron spin polarization and achieving a CH_4 generation rate 28.8 times higher than that of Co_3O_4 in CO_2 photoreduction.¹⁰⁰ Notably, the doping of Mn significantly suppressed the recombination of photogenerated electrons and holes by aligning their spin directions antiparallel (Fig. 13c). Mechanistic analysis reveals

that Mn sites enriched with holes initiate direct hydrogen transfer from H_2O , while adjacent Co sites preferentially capture electrons to activate CO_2 and convert it into $^*\text{COOH}$ intermediates.

Compared with metal element doping, non-metallic element doping, such as C, N, and B, typically reduces the overall material cost and minimizes environmental pollution risks, making it a low-cost and eco-friendly strategy. Wan *et al.* utilized $\text{O-ZnIn}_2\text{S}_4/\text{Ni}_{12}\text{P}_5$ to create a polarization-induced internal electric field through O doping and the formation of ohmic junctions, resulting in an 8.16-fold increase in photocatalytic H_2 generation compared to the pristine ZnIn_2S_4 .¹⁰⁴ Herein, O doping induced electron spin polarization, while the ohmic junctions enabled rapid migration of photogenerated electrons to the Ni_{12}P_5 active sites, effectively suppressing the recombination of photogenerated electron-hole pairs. TRPL decay spectra reveal that the average lifetime of photogenerated carriers in $\text{O-ZnIn}_2\text{S}_4/\text{Ni}_{12}\text{P}_5$ (2.58 ns) was significantly longer than that of ZnIn_2S_4 (1.01 ns), $\text{O-ZnIn}_2\text{S}_4$ (1.53 ns), and $\text{ZnIn}_2\text{S}_4/\text{Ni}_{12}\text{P}_5$ (2.21 ns). Additionally, the preferential dehydrogenation of the α -C-H bond in benzyl alcohol further facilitated hole transport and charge separation, ultimately enhancing both H_2 generation and benzaldehyde synthesis.

In another work, 2D CN nanosheets were fluorinated and then thermally defluorinated in Se vapor to prepare atomically thin Se-CN nanosheets.¹⁰⁸ Notably, Se doping expanded the light absorption edge of CN from 416 to 584 nm (bandgap reduced from 3.06 to 2.73 eV). In addition, the electron spin polarization in Se-CN significantly improved charge separation efficiency and surface catalytic reactions, which enhanced the

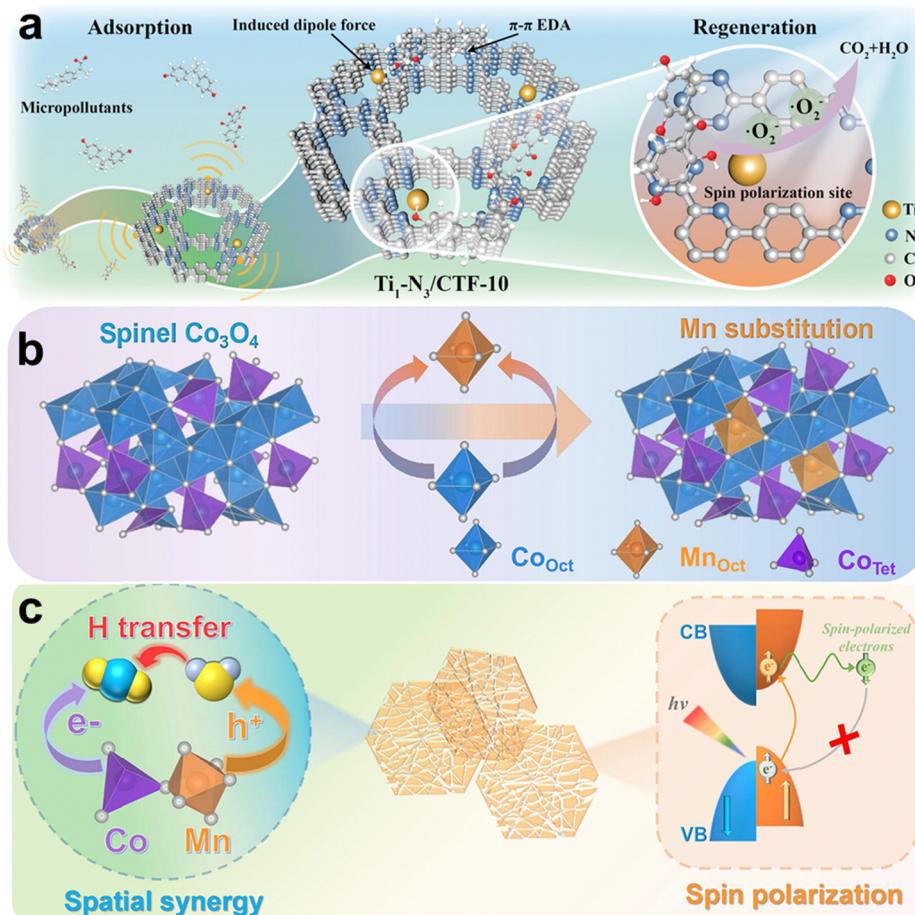


Fig. 13 Photocatalytic electron spin control by metal doping. (a) Schematic illustration of application scheme on spin polarization-related Ti-CTF for micropollutants removal. Reproduced with permission from ref. 99. Copyright 2023 John Wiley & Sons. (b) The schematically illustration of substitute Co by Mn. (c) Photocatalytic CO₂ methanation mechanism of Mn-Co₃O₄. Reproduced with permission from ref. 100. Copyright 2024 American Chemical Society.

performance of photocatalytic water splitting by 27 times compared to the pristine CN. Furthermore, Wang *et al.* designed 2D P/N-CN nanosheets through co-doping with P and N, leading to an optimized photocatalytic water splitting system that exhibited 200 times higher hydrogen production compared to CN, while enabling the selective oxidation of benzylamine *via* holes.¹⁰⁹ The spin-orbit coupling between N 2p and P 2p orbitals significantly increased the parallel spin arrangement, thereby accelerating the charge separation and surface catalysis in P/N-CN.

4.2 Defect engineering

It is commonly understood that introducing or adjusting defects (*e.g.*, defect type, concentration, and distribution) in photocatalysts can enhance the light absorption range and charge separation efficiency.³⁰ Notably, defect engineering also plays an important role in manipulating electron spin by modifying the local magnetic environment within the material, which can further influence the photocatalytic performance.^{13,23} Specifically, when defects are introduced into photocatalysts, they can create unsaturated atomic bonds or incomplete electron pairs

within the material's lattice, leading to the presence of unpaired electrons. These unpaired electrons can regulate the local magnetic moments, thereby causing spin polarization.

Metal defects can enhance the light absorption range, promote charge separation, and ultimately improve the catalytic activity of photocatalysts by modulating the electron spin structure around the metal atoms near the defects. For example, Zhang *et al.* prepared Ti-defective Bi₄Ti₃O₁₂ for the activation of O₂ to degrade tetracycline (TC) in water, which achieved a degradation rate constant 3.3 times higher than that of the pristine Bi₄Ti₃O₁₂.²³ The Ti defects on Bi₄Ti₃O₁₂ induced electron spin polarization, reducing the recombination of photogenerated electron-hole pairs. Additionally, surface Ti defects formed a center for adsorbing O₂ and extracting electrons, effectively generating HO[•], O₂^{•-}, and ¹O₂ for TC degradation. In another work, ZnO with abundant Zn defects was synthesized to optimize the spin structure and induce spin polarization by adjusting the electron occupancy of the e_g orbital, enabling simultaneous H₂ production and pollutant degradation.¹¹¹ Compared to the pristine ZnO, the Zn-defective ZnO achieved a 56.4-fold increase in H₂ production



rate and a 27.5-fold improvement in pollutant degradation efficiency.

In photocatalytic defect engineering, non-metallic defects are often preferred due to their ease of introduction through simple synthesis methods and their lower impact on the crystal structure, which preserves the photocatalyst's properties. Yang *et al.* employed WO_{3-x} nanowires rich in O defects to convert highly mobile and toxic U^{VI} into low-solubility and less toxic U^{IV} , achieving 79.9% reduction and immobilization of U^{VI} (initial concentration of 10 mg L^{-1}) at pH 5.¹¹² Compared to pristine WO_3 , the band gap of WO_{3-x} was reduced from 2.67 to 2.54 eV, enhancing its visible light absorption capacity (Fig. 14a). Additionally, the TRPL average lifetime decreased from 99.1 to 78.4 ns, which indicates the facilitated migration of electrons to the WO_{3-x} surface for U^{VI} reduction (Fig. 14b). In partial density of state (PDOS) spectra, the broader green region in WO_{3-x} compared to WO_3 suggests the presence of more O dangling bonds, promoting the spin polarization of W 5d band electrons (Fig. 14c). Moreover, density of state (DOS) spectra reveal that the CB of WO_{3-x} shifted to a lower energy region compared to WO_3 , which indicates an increased electron density in WO_{3-x} (Fig. 14d). Notably, the spin polarization in WO_{3-x} facilitated the separation of electrons and holes, while

the O defects created abundant active sites for photocatalytic reduction reactions (Fig. 14e). Together, these factors worked synergistically to enhance photocatalytic performance.

In another study, S defects were introduced into CdS, inducing a spin polarization effect that accelerated carrier transfer dynamics from the bulk phase to surface redox sites for photocatalytic water splitting.¹¹³ Following the introduction of S defects, the band gap of CdS decreased from 2.39 to 2.13 eV, while the TRPL average lifetime extended from 6.4 to 17.3 ns. This enhancement in light absorption and carrier separation efficiency resulted in a 50-fold increase in H_2 production compared to pristine CdS. Theoretical calculations reveal that H_2O molecules tend to adsorb onto Cd atoms adjacent to S defects. Subsequently, the $-\text{H}$ group in H_2O is reduced to H_2 , along with another $-\text{H}$ group from an adjacent adsorbed H_2O molecule. Simultaneously, O atoms occupy the S defects, forming Cd–O bonds, after which the two adjacent O atoms are oxidized to O_2 . Therefore, the spin polarization induced by S defects accelerated the separation of photogenerated charge carriers and enhanced the kinetics of surface photocatalytic reactions, effectively facilitating the overall H_2O splitting.

Recently, the introduction of dual-element defects in photocatalysts has been found to be effective in modulating the

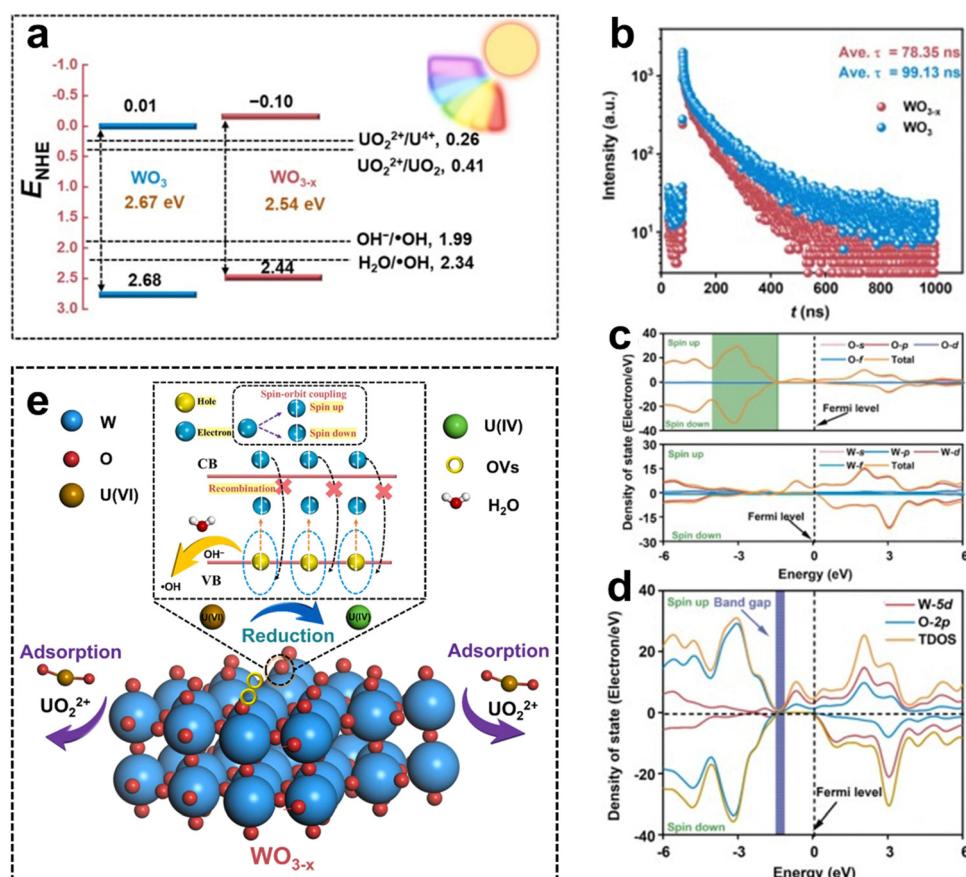


Fig. 14 O defect-induced spin polarization in WO_{3-x} for photocatalytic reduction and immobilization of $\text{U}(\text{VI})$. (a) Band structures and (b) TRPL spectra of WO_{3-x} and WO_3 . (c) PDOS and (d) DOS spectra of WO_{3-x} . (e) The schematic diagram on removal of $\text{U}(\text{VI})$ by WO_{3-x} through adsorption and photocatalytic reduction. Reproduced with permission from ref. 112. Copyright 2023 Elsevier.



material's electronic structure and enhancing the surface reaction activity.^{116,117} Additionally, the interactions between dual-element defects may create a synergistic effect, further improving the photocatalytic performance. For instance, Qi *et al.* introduced Cd and S defects on CdS nanorods by heat treatment, enhancing the photocatalytic H₂ production performance.¹¹⁶ The interfacial and polarization electric fields generated by the defects improved TRPL average lifetime from 0.86 to 0.95 ns and promoted faster interfacial charge migration, thereby improving the photocatalytic hydrogen production rate, with an AQE 8.5 times higher than that of pristine CdS. Moreover, Xu *et al.* developed Ti-based metal-organic frameworks (MOF) with Ti and organic linker defects to simultaneously achieve photocatalytic water splitting and pollutant degradation.¹¹⁷ Mechanistic analysis reveals that defect-induced electron spin polarization can control the CB position, resulting in an upward shift of the d-band center. This enhanced the adsorption of *H and facilitated the H₂ production. Additionally, spin polarization mitigated the recombination of photogenerated electrons and holes within Ti-MOF, leading to obvious improvements in photocatalytic activities.

4.3 Magnetic field regulation

Compared to traditional methods for enhancing photocatalytic performance, applying an external magnetic field during photocatalytic reactions is regarded as an efficient, safe, convenient, and contactless technique that significantly improves photocatalytic efficiency. The core mechanism behind this approach involves the spin-orbit and magnetic interactions of electrons. Electrons possess spin angular momentum, meaning that each electron has an intrinsic magnetic moment, akin to a tiny magnet. When exposed to an external magnetic field, electrons in the photocatalyst undergo spin-orbit and magnetic interactions, leading to adjustments in their spin states based on the field's direction and intensity.¹⁰ The spin-orbit interaction couples the electrons' spin with their orbital motion, while the magnetic interaction aligns the electrons' magnetic moments with the external field. This combined effect not only reconfigures the electrons' arrangement but also breaks the degeneracy of energy levels, causing them to split, which is known as the Zeeman effect. Specifically, the spin states divide into two levels: "parallel" (with spin aligned to the magnetic field) and "antiparallel" (with spin opposite to the magnetic field). Electrons preferentially occupy the lower-energy parallel state, resulting in spin polarization, where more electrons adopt the same spin orientation.¹³⁸

Additionally, the external magnetic field influences photo-generated charge carriers *via* the Lorentz force. The Lorentz force acts on moving charges in a magnetic field, effectively aiding in the separation and transfer of electron-hole pairs. This is particularly useful in magnetic materials, where the Lorentz force prevents their recombination and enhances photocatalytic efficiency. Moreover, the Lorentz force helps reduce catalyst particle agglomeration and increases the exposure of active sites, further enhancing the catalytic reaction.¹⁹

Li *et al.* found that applying a magnetic field enhanced the pollutant degradation efficiency of α -Fe₂O₃/reduced graphene

oxide (rGO).¹¹⁸ This improvement is attributed to the magnetic field facilitating carrier transfer from α -Fe₂O₃ to rGO, thereby greatly increasing carrier transfer efficiency. Moreover, it is reported that the application of an external magnetic field can control electron spin polarization in Mn–CsPbBr₃, which enhanced the efficiency of photocatalytic CO₂ reduction reactions.²⁰ When exposed to a 300 mT magnetic field or a permanent magnet, the photocatalytic performance of Mn–CsPbBr₃ improved by 5.7 times compared to pristine CsPbBr₃. This is because of the synergistic effect of doping with magnetic elements and applying a magnetic field, which increased the number of spin-polarized carriers and extended the carrier lifetimes.

In another study, applying an external magnetic field was found to manipulate spin polarization in Co_{3-x}O₄, achieving 100% selectivity for CO production during the photocatalytic reduction of CO₂.¹²⁶ Notably, the intrinsic spin polarization of Co_{3-x}O₄ (Fig. 15a), combined with the antipodal Lorentz forces (Fig. 15b), enhanced the TRPL average lifetime from 0.14 to 3.51 ns, significantly boosting the charge carrier mobility. Additionally, spin polarization induced by Co defects and the external magnetic field provided a thermodynamic advantage for CO₂ reduction by facilitating the conversion of CO₂ to *COOH and subsequently to *CO, while lowering the free energy required for the reduction reaction (Fig. 15c and d).

In addition, controlling charge separation by the combination of ferroelectric polarization and spin polarization has recently gained attention as a promising approach to boost the efficiency of photocatalytic reactions. For instance, Chiang *et al.* demonstrated the modulation of charge separation in 2D ferroelectric CuInP₂S₆ crystals through the ferroelectric field and an applied magnetic field, which enhanced the photocatalytic reduction of CO₂.¹²⁸ Specifically, the ferroelectric polarization of CuInP₂S₆ can be regulated through ferroelectric phase transitions and electric polarization, while the electron spin polarization can be tuned by introducing S defects and applying an external magnetic field, enabling the simultaneous manipulation of both ferroelectric and spin polarizations, which facilitated the conversion of CO₂ to CO and CH₄. The *in situ* diffuse reflectance infrared Fourier transformed (DRIFT) spectra exhibit the formation of various surface adsorbed species during the photocatalytic CO₂ reduction, including *COOH, *CH₃O, and *CHO, which are all key intermediates for the conversion of CO₂ to CH₄. Notably, compared to pristine CuInP₂S₆, the increased DRIFT intensity in CuInP₂S₆ with S defects indicates enhanced CH₄ production, which is attributed to the promotion of charge separation through both ferroelectric and spin polarizations.

4.4 Metal coordination modulation

Modulating the metal coordination environment in a photocatalyst can typically alter the energy level distribution of the metal center and the local electron cloud around the metal, thereby manipulating the spin state of the metal and inducing spin polarization.¹¹⁹ Current methods to adjust the metal coordination environment include controlling the number of



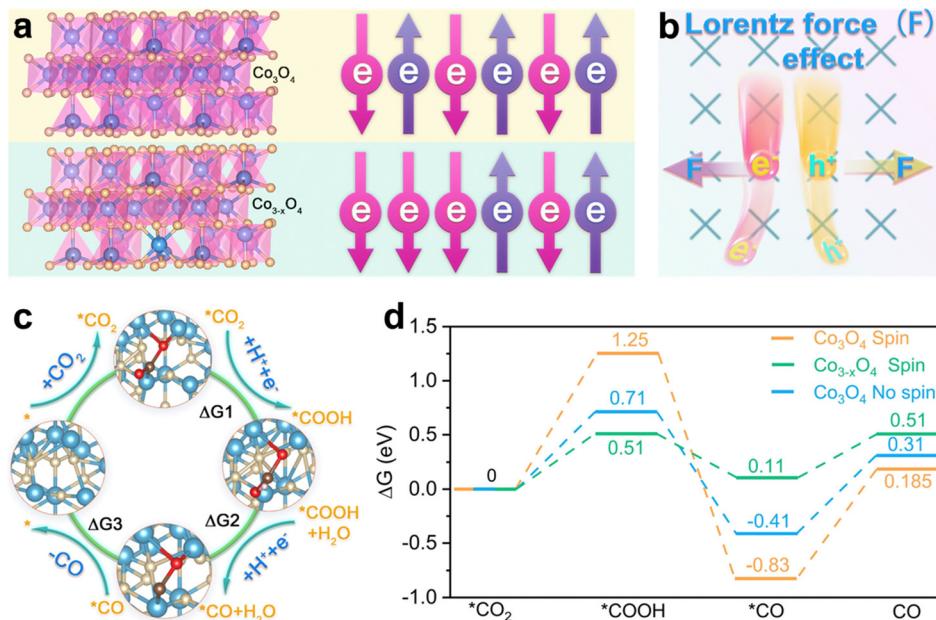


Fig. 15 Spin polarization-enhanced photocatalytic CO_2 reduction by $\text{Co}_{3-x}\text{O}_4$ under a magnetic field. (a) Crystal structure and electron spin distribution of Co_3O_4 and $\text{Co}_{3-x}\text{O}_4$. (b) Schematic diagram of the Lorentz force effect on photogenerated carrier transfer. (c) Proposed pathway for CO_2 photoreduction to CO . (d) Calculated free energy diagrams of photoreduction of CO_2 to CO on Co_3O_4 and $\text{Co}_{3-x}\text{O}_4$ with and without consideration of spin polarization in the calculation. Reproduced with permission from ref. 126. Copyright 2024 American Chemical Society.

ligands around the metal, altering the ligand types at the metal center, and modifying the geometric symmetry of the coordination environment (e.g., octahedral and tetrahedral).¹³⁹ For instance, Sun *et al.* synthesized three different Co-doped Zn-based MOFs, with Co centers coordinated to $-\text{CH}_3\text{COO}$ (Co-OAc), $-\text{Br}$ (Co-Br), and $-\text{CN}$ (Co-CN), by varying the Co precursor (Fig. 16a).¹¹⁹ It was proposed that during the photocatalytic CO_2 reduction process, spin-up electrons are excited from the HOMO to the LUMO under visible light irradiation, while spin-up holes remain in the HOMO. Due to spin-orbit coupling (SOC) and hyperfine interactions (HFI), the excited spin-up electrons transition to a spin-down, while the spin-up holes retain their spin orientation. This spin mismatch between electrons and holes effectively suppresses their recombination (Fig. 16b). Both experimental results and DFT calculations reveal that modulating the coordination environment of the Co site adjusted its spin state, resulting in varying levels of charge separation efficiency and CO_2 reduction capacity in Co-OAc, Co-Br, and Co-CN. Notably, Co-OAc, with its highest spin state, exhibited exceptional charge separation and optimized CO_2 adsorption and activation energy barrier (Fig. 16c), leading to a remarkable photocatalytic CO_2 reduction rate of $2.3 \text{ mmol g}^{-1} \text{ h}^{-1}$ and a 99.1% selectivity for CO production.

Additionally, Huang *et al.* proposed to regulate the spin state of the Fe center in the Fe-Pt Hofmann clathrates through facilitating the valence state conversion of Pt coordinated with Fe from Pt^{IV} to Pt^{II} by iodine treatment.¹²⁰ Notably, when the spin state of Fe^{II} in the Fe-Pt Hofmann clathrates is high spin, only the photocatalytic oxygen reduction reaction (ORR) occurs, leading to the synthesis of H_2O_2 . Conversely, when Fe^{II} is in a low spin state, both ORR and water oxidation reactions (WOR)

can take place. This spin state transition enables the switching between ORR and WOR in H_2O_2 photosynthesis (Fig. 17a). This is because the low spin state of Fe^{II} reduced the energy barriers for both ORR and WOR processes. Moreover, compared to the high spin state, the low spin state of Fe^{II} exhibited enhanced light harvesting ability, improved photogenerated carrier separation efficiency, and superior charge transfer capability.

In another study, the spin state of Fe^{III} in a $\text{Au}_3\text{Fe}_1/\text{Mo}$ alloy photocatalyst can be finely tuned to enhance photocatalytic N_2 fixation by altering the metal coordination environment.⁴⁹ Theoretical calculations indicate that the strong electronic interactions among Mo, Fe, and Au enable the Fe site to serve as a crucial center for N_2 adsorption and activation. Specifically, the unpaired electrons of N_2 first occupy the empty d orbital of Fe^{III} , after which the unpaired electrons from Fe^{III} fill the antibonding orbital of $\text{N}\equiv\text{N}$, resulting in its dissociation. In high spin Fe sites, the lack of an empty orbital hinders N_2 activation. In contrast, medium spin Fe sites have an empty orbital that can accept these unpaired electrons from N_2 , which then feed back into the $\text{N}\equiv\text{N}$ antibonding orbital, weakening the $\text{N}\equiv\text{N}$ bond and facilitating the N_2 fixation process (Fig. 17b).

4.5 Chiral induced spin selectivity

The chiral induced spin selectivity (CISS) effect refers to the preferential alignment of electron spins in a specific direction within a chiral molecule (whose structure cannot be superimposed on its mirror images) after charge polarization (Fig. 18a), which plays a crucial role in controlling chemical reactions, enantiomer separation, and biorecognition processes.^{16,140} When an electron moves through a chiral molecule,



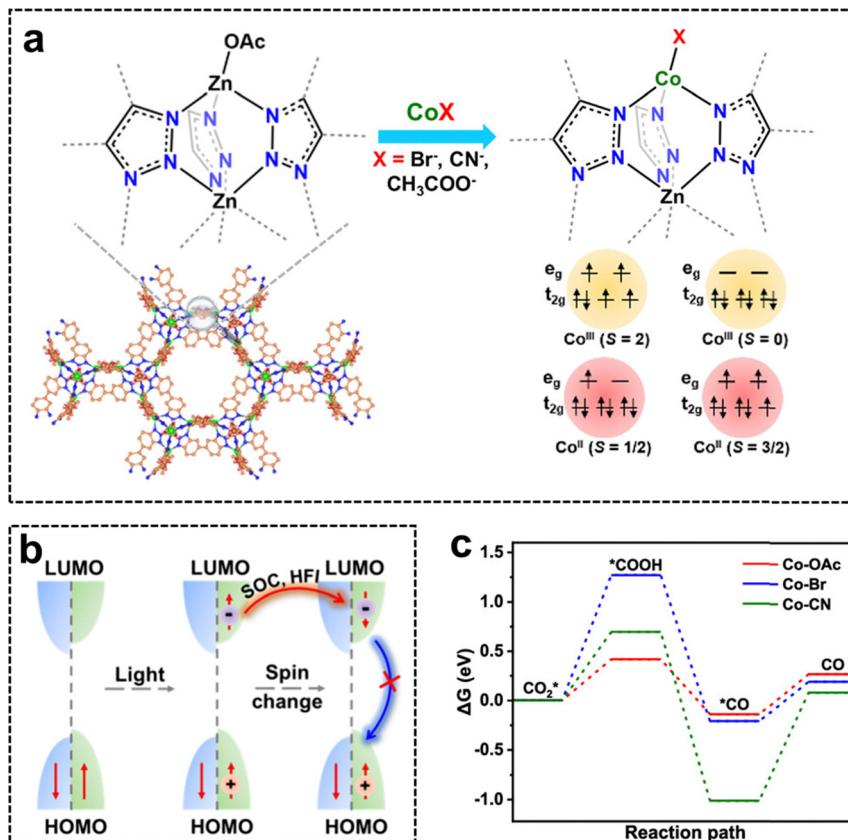


Fig. 16 Manipulating the coordination environment of Co sites for boosting CO_2 photoreduction. (a) Synthesis of Co–OAc, Co–Br, and Co–CN featuring different spin states of Co^{II} and Co^{III} species via a postsynthetic exchange strategy. (b) Mechanism of spin polarization for improved charge separation. (c) Energy variations of CO_2 photoreduction along the reaction path of the three Co-based MOFs. Reproduced with permission from ref. 119. Copyright 2024 American Chemical Society.

the curvature of the potential energy within the chiral structure generates a centripetal force that acts perpendicular to the electron's velocity (Fig. 18b).¹⁶ The direction of this force depends on the handedness (left- or right-handed) of the chiral molecule. Moreover, the centripetal force can create an effective magnetic field (B) along the electron's path, which interacts with the electron's magnetic moment, stabilizing one spin orientation while destabilizing the other. This preferred spin orientation is determined by the chiral axis of the chiral molecule, which is influenced by its handedness and linear momentum. Additionally, when a chiral molecule is subjected to an electric field along its axis, charge polarization occurs, accompanied by spin polarization ($\delta^{(+)}$ and $\delta^{(-)}$ at the ends of the helix) (Fig. 18c).¹⁶ Notably, the spin polarization associated with each pole is influenced by the handedness of the chiral molecule. Thus, when a chiral molecule participates in an electron transfer process, it can selectively facilitate or inhibit electron transfer based on the spin direction of the electrons. In other words, the CISS effect enables chiral molecules to act as spin filters, allowing only electrons with a specific spin orientation to pass while blocking others (Fig. 18d). In addition, the spin-orbit coupling (SOC) model posits that the spin of electrons within a chiral molecule is intertwined with its orbital motion, linking the CISS effect to the intricate interaction

between electron spin and molecular chirality (Fig. 18e).¹⁴⁰ As a result, the electron spin is closely tied to the molecular chirality.

Mott polarimetry is a powerful technique for investigating the magnetic properties of thin films and characterizing the CISS effect. In this approach, electrons striking an Au foil with significant spin-orbit coupling are scattered at varying angles based on their spin orientation, with the angle-dependent detection providing quantitative data about the electron spin population (Fig. 18f).¹⁴¹ Specifically, the typical procedure involves first exciting photoelectrons in the substrate (i), which then pass through a chiral spin filter (ii), resulting in spin polarization. These photoelectrons are subsequently scattered onto the Au foil according to their spin orientation (iii) and quantified using two independent detectors (iv). Additionally, the CISS effect is often demonstrated using magnetic conductive probe atomic force microscopy (mCP-AFM), which combines AFM with magnetic conductivity measurements.¹⁴² This technique assesses a material's conductivity and magnetic properties by applying an external magnetic field during the scanning process. The structure of chiral molecules modulates the electron spin during migration, resulting in spin-selective charge transfer. Using mCP-AFM, researchers can quantify the strength of the CISS effect and elucidate how chiral molecules influence electron transfer.



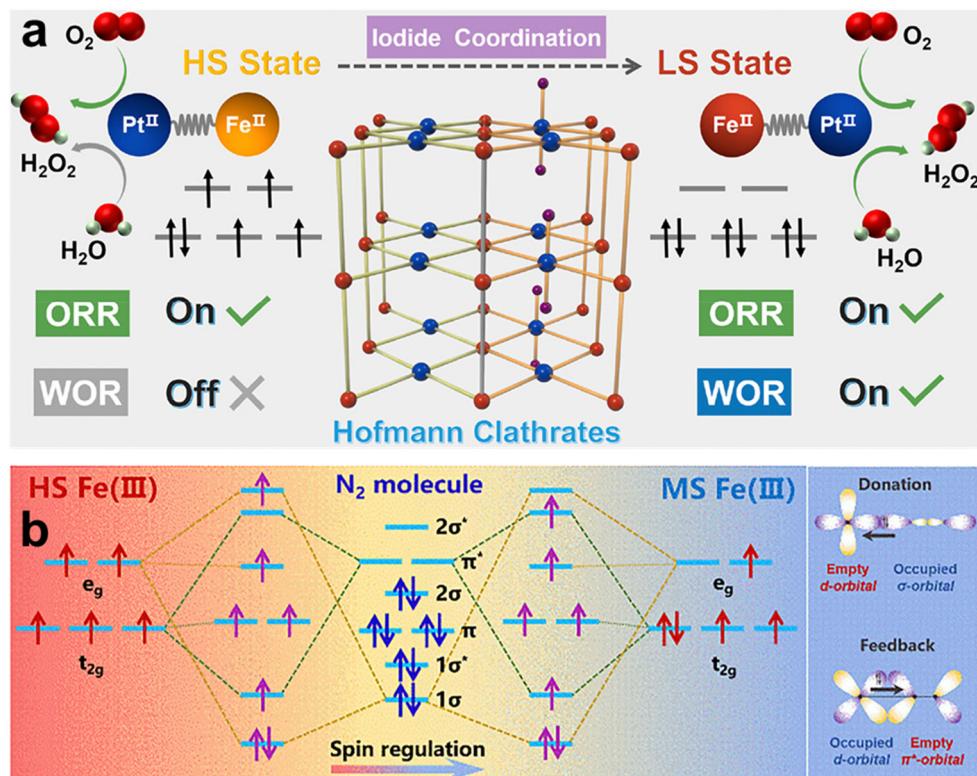


Fig. 17 Modulation of metal coordination environment for electron spin control. (a) Schematic illustrations of the dynamic spin-state transition for Fe^{II} in the Hofmann clathrates and the spin-related photocatalytic H_2O_2 synthesis overall reaction. Reproduced with permission from ref. 120. Copyright 2023 American Chemical Society. (b) Schematic of orbital interaction between the metal and N_2 in medium-spin $\text{Au}_3\text{Fe}_1/\text{Mo}$. Reproduced with permission from ref. 49. Copyright 2024 Elsevier.

Chiral materials encompass both organic materials, such as chiral polymers, chiral MOFs, and chiral quantum dots, as well as inorganic materials, including chiral oxides, chiral metals, and chiral silicon materials. Inorganic chiral materials typically exhibit superior thermal stability, mechanical strength, and optical properties, making them suitable for a broader range of applications, including photocatalysis, optoelectronic devices, and sensors.¹⁴³ Additionally, the synthesis of inorganic materials often allows for more precise control over their crystal structure and chiral characteristics, enhancing their overall performance and functionality. Chirality in inorganic nanomaterials is intricately related to their shape, crystal structure, and interactions with chiral ligands. The origins of chirality in these materials can be categorized into four main types (Fig. 19): (a) intrinsic chirality from chiral crystals;^{144–146} (b) chiral interactions between inorganic nanomaterials and chiral ligands;^{147–150} (c) chiral shapes at sub-wavelength scales;^{151–153} and (d) chiral assemblies formed by inorganic nanoparticles.^{154–157} Intrinsic chirality in chiral crystals originates from the asymmetric arrangement of atoms within their lattice, primarily involving Sohncke space groups (such as rotation, rotational translation, and translation), as well as lattice distortions or defects (Fig. 19a).^{144–146} When synthesized under appropriate conditions, nanocrystals with chiral space groups can develop chiral lattices featuring uniform rotational stacking. A robust method for introducing chirality into inorganic

nanomaterials is interacting with chiral ligands or molecules, such as deoxyribonucleic acid (DNA) and amino acids, which can induce chirality in nearby achiral materials (Fig. 19b).^{147–150} This can be achieved either by direct synthesis in the presence of chiral ligands or through post-synthesis ligand exchange, where initial achiral ligands are replaced by chiral ones. Chiral-shaped inorganic nanomaterials, characterized by mirror-asymmetric geometries, are typically synthesized using two main methods: bottom-up and top-down (Fig. 19c).^{151–153} In the bottom-up approach, chiral ligands direct atoms to form chiral stacking patterns, enabling the synthesis of chiral nanostructures from various inorganic materials. Top-down methods, on the other hand, use advanced tools such as focused ion beams, electron beam lithography, and multi-photon direct laser writing to precisely shape chiral nanostructures, allowing for meticulous control over their size, geometry, and symmetry. However, the top-down methods typically come with high equipment costs and complex processes. Chiral assembly refers to the organization of inorganic nanomaterials into chiral configurations, typically facilitated by chiral ligands and polymer composites acting as scaffolds or bridges (Fig. 19d).^{154–157} These ligands promote interactions between nanomaterials, leading to their chiral arrangement. Furthermore, the chiral ligands on the surface of inorganic nanomaterials can mutually attract, resulting in the formation of chiral superstructures.

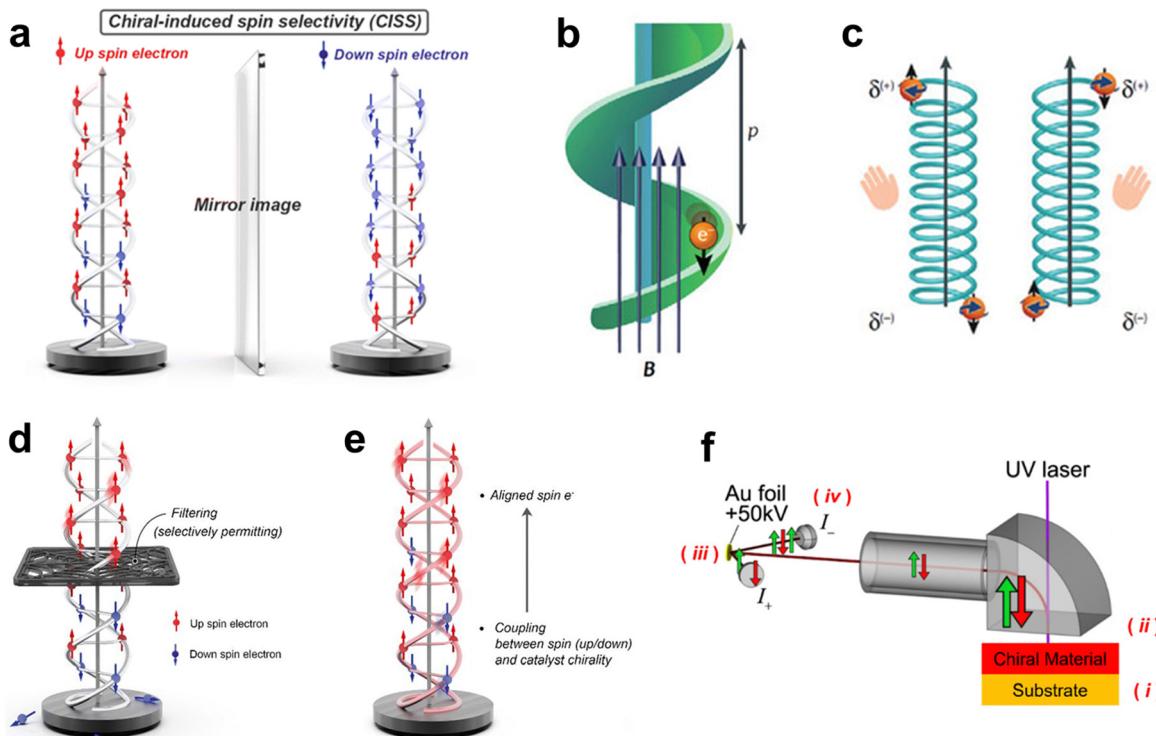


Fig. 18 Fundamentals of CISS effect. (a) Schematic of CISS effect in the transport of electrons with opposite spin through the chiral structure. Reproduced with permission from ref. 140. Copyright 2024 Royal Society of Chemistry. (b) Schematic of the electron transmission through a chiral potential of pitch p . (c) Schematic of the charge and spin polarization in chiral molecules, when the molecules are exposed to an electric field acting along their axes. Reproduced with permission from ref. 16. Copyright 2019 Springer Nature. (d) The spin-filter model demonstrating how electrons with specific spin states are selectively transmitted. (e) The SOC model highlighting the interaction between an electron's spin and its orbital motion within the chiral structure. Reproduced with permission from ref. 140. Copyright 2024 Royal Society of Chemistry. (f) Schematic of CISS determination using Mott polarimetry measurements. Reproduced with permission from ref. 141. Copyright 2024 American Chemical Society.

Recent advancements in the CISS effect have demonstrated its significant potential for controlling the spin orientation of electrons, providing a novel approach to enhancing the performance of photocatalytic water splitting. In this process, the oxygen evolution reaction is usually regarded as the rate-limiting step, as the oxidation of H_2O by photogenerated holes generates HO^\bullet with random spin orientations. These free radicals with antiparallel spin orientations tend to combine into the singlet byproduct H_2O_2 rather than forming triplet O_2 . The CISS effect can align the unpaired electrons in HO^\bullet in a parallel configuration, facilitating their combination along the triplet pathway to produce O_2 .¹⁴⁰ For example, Pan *et al.* synthesized chiral ZnO to induce the CISS effect in photocatalytic water splitting, offering a robust strategy for manipulating electron spin-dependent redox reactions.¹² Magnetic circular dichroism (CD) spectroscopy reveals that the chiral-structured ZnO functioned as a spin filter, inducing the CISS effect in the photoinduced carriers (Fig. 20a). The photocatalytic water splitting performance of chiral L- and D-ZnO was notably higher than that of the achiral DL-ZnO, with activities 2.0 and 1.9 times greater in photocatalytic O_2 production, respectively (Fig. 20b). Moreover, the charge carrier dynamics of chiral and achiral ZnO were investigated using TAS, with the data analyzed using a triple exponential decay function to account for electron

capture. The average carrier lifetimes for L- and D-ZnO were determined to be 4.2 and 2.1 ns, respectively, representing increases of 2.4 and 1.2 times compared to DL-ZnO (Fig. 20c). This extended carrier lifetime is attributed to the high spin polarization induced by the CISS effect (Fig. 20d). Mechanistic analysis indicates that polarized holes regulated the spin alignment of HO^\bullet intermediates, preventing the formation of H_2O_2 from spin-antiparallel HO^\bullet recombination and ultimately enhancing the efficiency of O_2 production (Fig. 20e).

4.6 Combined strategies

Generally, combining two electron spin control strategies can generate a synergistic effect, optimizing light absorption, electron-hole pair separation efficiency, and surface reaction kinetics, resulting in a comprehensive enhancement of photocatalysis. For example, Li *et al.* utilized an external magnetic field along with single-atom Co doping to synergistically enhance the electron spin polarization of C_3N_4 , promoting photocatalytic water splitting and the simultaneous oxidation of benzylamine.¹²⁴ With a modest external magnetic field of 24.5 mT, the H_2 production rate of Co- C_3N_4 reached approximately 340 times that of pristine C_3N_4 in the absence of a magnetic field. Experimental results and theoretical calculations reveal that the interaction between Co d and N p orbitals



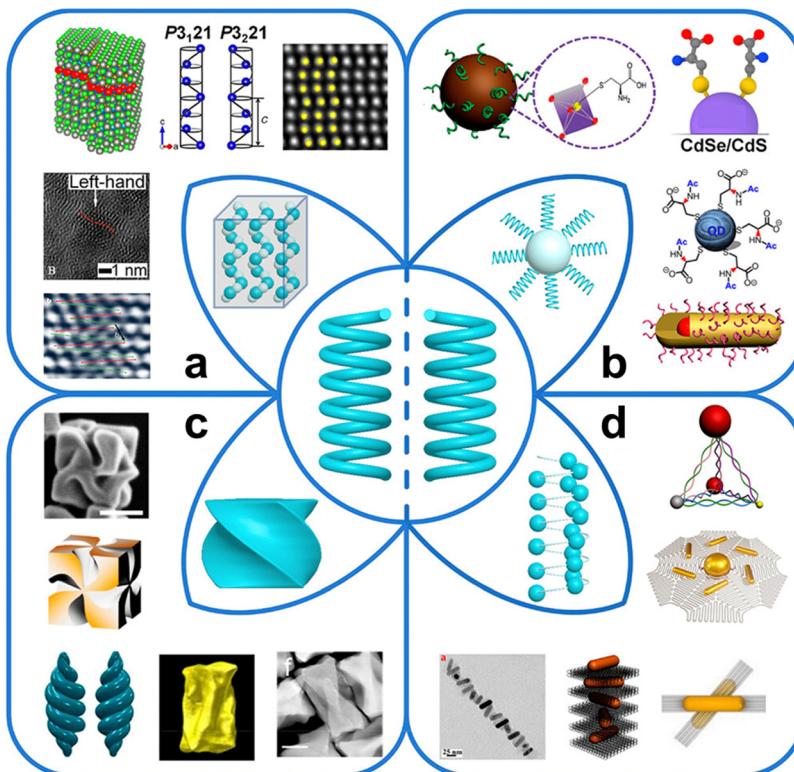


Fig. 19 Schematic diagram of chirality origin in inorganic nanomaterials. Reproduced with permission from ref. 143. Copyright 2023 American Chemical Society. (a) Intrinsic chirality from chiral crystals. Reproduced with permission from ref. 144. Copyright 2015 American Chemical Society. Reproduced with permission from ref. 145. Copyright 2020 Springer Nature. Reproduced with permission from ref. 146. Copyright 2013 Springer Nature. (b) Chiral interactions between inorganic nanomaterials and chiral ligands. Reproduced with permission from ref. 147. Copyright 2016 American Chemical Society. Reproduced with permission from ref. 148. Copyright 2018 American Chemical Society. Reproduced with permission from ref. 149. Copyright 2018 John Wiley and Sons. Reproduced with permission from ref. 150. Copyright 2019 American Chemical Society. (c) Chiral shapes at subwavelength scales. Reproduced with permission from ref. 151. Copyright 2018 Springer Nature. Reproduced with permission from ref. 152. Copyright 2020 Springer Nature. Reproduced with permission from ref. 153. Copyright 2022 John Wiley and Sons. (d) Chiral assemblies formed by inorganic nanoparticles. Reproduced with permission from ref. 154. Copyright 2012 American Chemical Society. Reproduced with permission from ref. 155. Copyright 2015 American Chemical Society. Reproduced with permission from ref. 156. Copyright 2014 Springer Nature. Reproduced with permission from ref. 157. Copyright 2012 American Chemical Society.

altered the symmetry center of C_3N_4 , leading to increases in both the dielectric constant and spin polarization.

Furthermore, the magnetic field enhanced the alignment of electron spins, increasing the number of electrons with parallel spins. Thus, the combination of the magnetic field and Co doping strategy created a significant built-in electric field and active sites, which improved charge transfer and surface reactions. In another study, a dilute magnetic Ni–CdS/MoS₂ photocatalyst with high ferromagnetic spin polarization was developed by doping Ni ions into CdS/MoS₂ for photocatalytic water splitting.¹²⁵ The electron spin polarization induced by Ni doping and the applied magnetic field reduced charge recombination in Ni–CdS/MoS₂ and improved the interfacial transfer efficiency between CdS and MoS₂, resulting in a 3.89-fold increase in the photocatalytic H₂ production rate.

The combination of defect engineering and magnetic field regulation also exhibits synergistic electron spin control for boosting photocatalytic activities. For instance, BaTiO₃ with O defects exhibited remarkable activity under an applied magnetic field, achieving an NH₃ yield of over 1.93 mg L⁻¹ h⁻¹ in

photocatalytic N₂ fixation.⁵⁰ The preparation of O defect-modified BaTiO₃ involves mechanically grinding BaTiO₃ with NaBH₄ in a glove box, followed by reducing the mixture with molten NaBH₄ in an Ar atmosphere to create O defects on the BaTiO₃ surface. Characterization tests reveal that O defects not only induced lattice distortion and valence state changes of BaTiO₃ but also increased local disorder, facilitating the spontaneous polarization of surface electrons. DFT calculations indicate that the desorption of NH₃ to form *N is the rate-determining step in the photocatalytic N₂ fixation process. Interestingly, the internal electric field induced by O defects and the applied magnetic fields reduced the energy barrier of this step, thereby accelerating NH₃ production. Thus, the electromagnetic synergy between the internal electric field and the external magnetic field was achieved by controlling O defects, effectively suppressing the recombination of photo-generated charge carriers and promoting the reduction of N₂.

Besides using magnetic fields, combining doping with defect engineering is also an effective strategy for electron spin control. For example, Ni–CdS photocatalyst was synthesized by



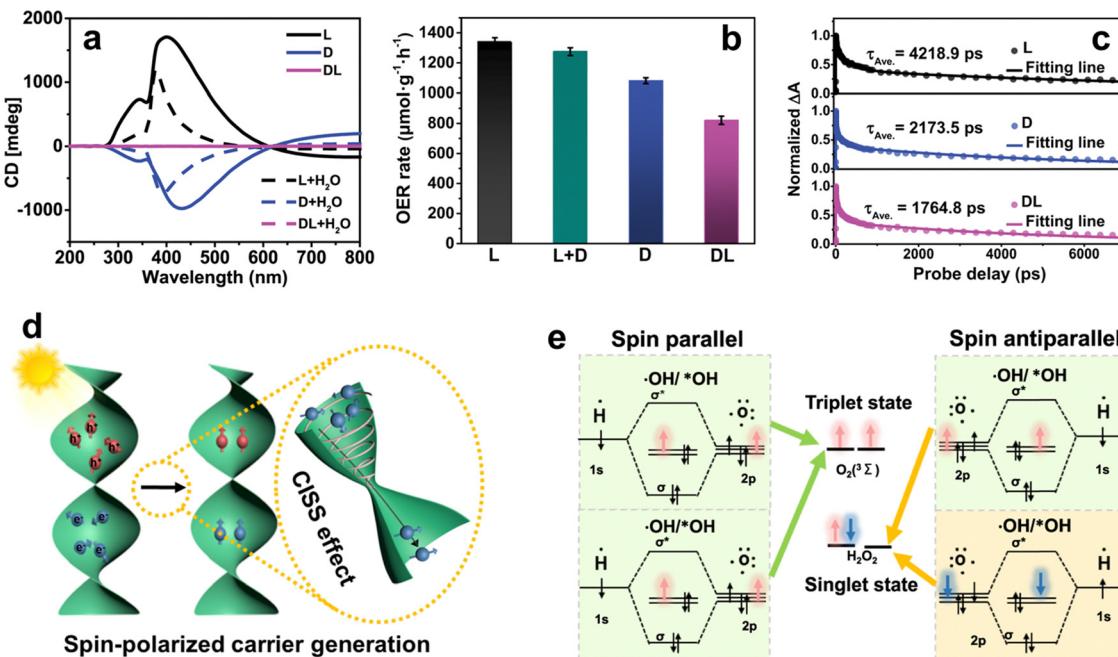


Fig. 20 CISS effect in chiral ZnO for photocatalysis. (a) CD spectra of ZnO and after infiltration with water. (b) Photocatalytic oxygen production of powdered photocatalysts. (c) Normalized carrier decay kinetics of L-, D-, and DL-ZnO. (d) Schematic diagram of spin selection of photoinduced carriers in chiral ZnO. (e) Formation of singlet or triplet products by HO[•] with different spin directions. Reproduced with permission from ref. 12. Copyright 2023 Springer Nature.

incorporating highly dispersed Ni atoms into the CdS lattice to replace Cd atoms.¹²⁹ This incorporation induced local lattice distortion and S defects, which increased the dipole moment and enhanced the spin-polarized electric field. The resulting charge redistribution, driven by the enhanced internal electric field, shifted the S p-band center downward, facilitating the desorption of the *H intermediate during photocatalytic water splitting, ultimately achieving a hydrogen yield 5.58 times higher than that of pristine CdS.

4.7 Other strategies

Recently, the development of van der Waals heterostructures has been shown to modify the electronic band structure and local electron density distribution in photocatalysts.^{131,132} These alterations influence the distribution of electron spin states, leading to spin polarization effects. By integrating heterostructures with spin polarization, it becomes possible to suppress the recombination of photogenerated charge carriers while enhancing the activity and selectivity of surface redox reactions. For example, spin-polarized monolayer graphene was *in situ* grown on 2D MoS₂ using chemical vapor deposition to form van der Waals heterostructures capable of photocatalytic overall water splitting under visible light.¹³¹ The spin polarization facilitated the spatial separation of photo-generated electrons and holes in MoS₂ and directed electrons to active sites on the graphene surface through heterojunctions, resulting in an extended TRPL average lifetime from 6.0 to 27.7 ns. Moreover, CN/graphene composites with van der Waals heterostructures and spin-polarized electronic properties have demonstrated high charge separation efficiency and enhanced

surface catalytic reactions, achieving efficient photocatalytic water splitting.¹³²

Altering the oxidation state of a transition metal can directly change the number of d-electrons, which in turn affects the orbital occupancy. As a result, the number of unpaired electrons varies, influencing the spin state. Gong *et al.* regulated the spin state of Co-COF by altering the oxidation state of Co at the porphyrin center, significantly promoting photocatalytic CO₂ reduction and enhancing selectivity for HCOOH production.⁶⁸ Theoretical calculations reveal that in Co^{III}-COF, the Co-3d_{z²} orbitals interact with the O-2p orbitals of CO₂, whereas in Co^{II}-COF, the Co-3d_{xz} or Co-3d_{yz} orbitals couple with the O-2p orbitals. Notably, the CO₂ adsorption energy in Co^{III}-COF is more than twice that in Co^{II}-COF. Moreover, Co^{III}-COF exhibited stronger interactions with HCOOH, further promoting selective HCOOH generation. These findings highlight that the spin state transition induced by metal valence regulation plays a crucial role in optimizing photocatalytic performance.

In addition, regulating the size of a nanocrystal can influence its spin state, primarily due to the enhanced quantum confinement effects that arise as the nanocrystal size decreases.¹⁵⁸ In smaller nanocrystals, discrete energy levels replace continuous ones, altering the distribution and alignment of electron spins. For instance, Fe₂O₃ nanoclusters with low spin Fe were synthesized through a nanocrystal size modification strategy and subsequently anchored on g-C₃N₄ for photocatalytic pollutant degradation.¹³³ Notably, the low spin Fe₂O₃/g-C₃N₄ exhibited superior photocatalytic activity compared to its high spin counterpart. This improvement was attributed to the d-band center of



the Fe 3d orbital in low spin Fe being closer to the Fermi energy level, which weakened the antibonding state. Consequently, the Fe–O interaction was strengthened, enhancing the production of HO^{\bullet} and $\text{O}_2^{\bullet-}$ for the degradation of diclofenac.

It has been reported that the pyroelectric field can induce the reconstruction of electron spin states.¹³⁴ He *et al.* prepared $\text{FeS}_2/\text{CuCo}_2\text{O}_4$ hollow core–shell heterojunctions and utilized temperature fluctuations between hot and cold regions to drive the pyroelectric field. Specifically, FeS_2 generated heat under NIR light, providing the necessary high-temperature endpoint for the pyroelectric field in CuCo_2O_4 . This pyroelectric field enabled CuCo_2O_4 to release surface charges and generate spontaneous polarization. Additionally, it modified the filling of electrons on Co sites and reversed the electron spin state. This resulted in an extension of TRPL average lifetime from 5.7 to 15.0 ns, enhancing photocatalytic water splitting with an average AQE of up to 19.8%.

5. Conclusions and outlook

Electron spin control of photocatalysts has emerged as a promising strategy for precisely manipulating the spin and spin state of electrons to optimize the separation of photogenerated electron–hole pairs and their subsequent reactions on the photocatalyst surface, offering an innovative approach to improve photocatalytic efficiency. By modulating the electron spin of active sites to enhance interactions with reactants and aligning the electron spin in a specific direction to improve product selectivity, this approach also offers a novel strategy for regulating surface reaction pathways. Summarizing recent progress in electron spin control within the field of photocatalysis is therefore of particular importance. This review provides a comprehensive overview of advanced strategies for manipulating electron spin, including doping design, defect engineering, magnetic field regulation, metal coordination modulation, CISS effect, and combined strategies, aiming at enhancing photocatalytic processes such as water splitting, CO_2 reduction, pollutant degradation, and N_2 fixation.

Among these strategies, doping design, defect engineering, magnetic field regulation, metal coordination modulation, and CISS effect are recognized as key approaches. We compared the number of published academic papers utilizing these strategies to promote photocatalytic applications, with the results presented in a pie chart in Fig. 21a. The findings indicate that doping design is the most widely employed strategy, while defect engineering and magnetic field regulation have also received considerable attention. In contrast, research on electron spin manipulation through metal coordination modulation and the CISS effect remains relatively limited.

A comparison of the five strategies based on multiple metrics, including light absorption, charge separation, surface reaction efficiency, selectivity, stability, and practicality, is presented in Fig. 21b, with a detailed overview of the advantages and limitations provided in Fig. 21c. Notably, each strategy has unique strengths and limitations that must be carefully

considered when designing spin control approaches. For instance, doping design remains the most popular strategy due to its ability to enhance light absorption, increase active sites, provide high stability, and offer a straightforward synthesis process. However, it also presents several challenges, including the potential formation of recombination centers that reduce the separation efficiency of photogenerated carriers. Additionally, structural defects introduced during doping can adversely impact the optical and electronic properties of photocatalysts. Furthermore, controlling the doping level is often difficult, ultimately affecting the reliability of photocatalytic performance. In contrast, defect engineering significantly enhances charge separation but introduces limitations, such as uncontrolled defect formation, structural instability, and reduced efficiency over time, making instability a key challenge to address. The application of an external magnetic field during photocatalytic reactions is an efficient and contactless technique that not only promotes charge separation but also improves reaction selectivity. Nevertheless, this approach requires specialized magnetic field generation equipment and precise control, inevitably increasing the complexity and cost of operation. Furthermore, not all photocatalysts are compatible with magnetic field, particularly flexible or non-magnetic materials. Additionally, the effectiveness of magnetic field can be sensitive to environmental factors such as temperature and humidity, potentially affecting the stability and repeatability of the photocatalytic reaction.

Currently, research on metal coordination modulation and the CISS effect remains relatively limited. This may be due to unclear mechanisms of spin control and the underdeveloped state of material preparation techniques, which complicate the implementation of these strategies. The challenges typically associated with metal coordination modulation and the CISS effect include complex photocatalyst synthesis, stability issues, and limited scalability. However, these two strategies also offer significant advantages. Metal coordination modulation allows for precise active site tuning and design flexibility, while the CISS effect promotes charge separation and improves reaction selectivity. These benefits are likely to attract increased research attention, driving the development of new photocatalysts that enable robust electron spin control. Thus, when designing photocatalysts and spin control strategies, it is essential to carefully consider both the advantages and limitations to select the most appropriate approach.

Despite the continuous progress made in recent years in electron spin control for optimizing photocatalysis, several areas still require further development and improvement:

1. Further improving strategies for electron spin control. Developing more robust manipulation techniques is essential to fully realize the potential of electron spin control and enhancing photocatalytic efficiency. For instance, advanced doping techniques such as co-doping can boost light absorption and improve spin polarization, while employing suitable non-metallic dopants reduces reliance on rare metals and minimizes ecological impact.^{109,110} Moreover, optimizing the concentration and distribution of dopants can lead to more



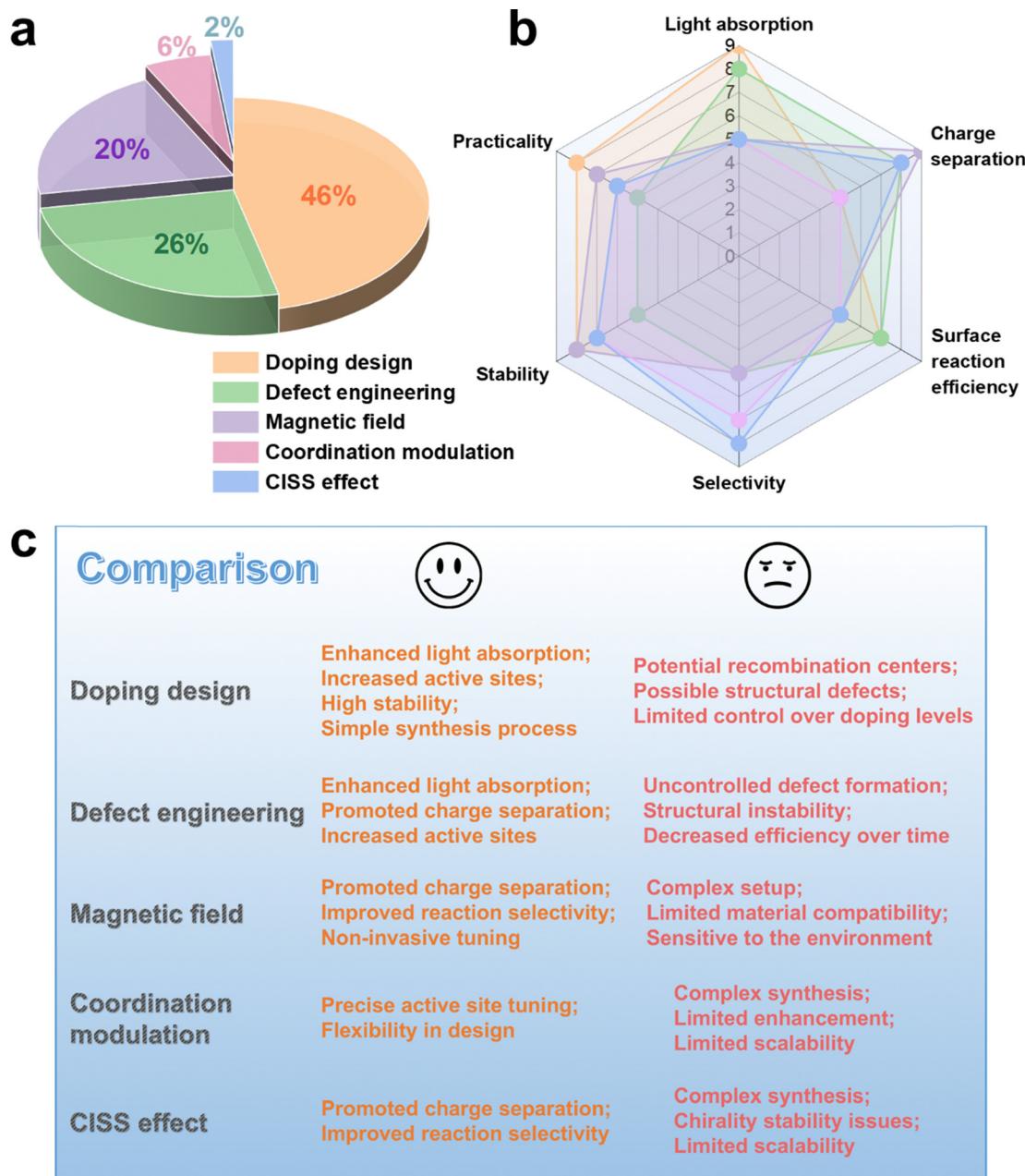


Fig. 21 Comparative analysis of doping design, defect engineering, magnetic field regulation, metal coordination modulation, and the CISS effect. (a) Percentage of publications employing the five strategies. (b) Comparison of the five strategies from multiple metrics. (c) Comparison of the advantages and limitations of the five strategies.

precise electron spin control, ultimately raising photocatalytic performance.

Employing defect engineering to create and control defects within the photocatalyst lattice can introduce local states that facilitate spin manipulation. Exploring the relationship between different defect types (*e.g.*, metallic and non-metallic defects) and their impacts on electron spin control is essential for enhancing photocatalytic performance. Additionally, developing defect engineering techniques that precisely regulate the location and concentration of defects, while also improving the stability of the synthesized materials, is

essential to ensure their long-term efficacy in photocatalytic applications.

Currently, most studies on magnetic fields in photocatalysis focus on static magnetic fields, where both the direction and intensity remain constant throughout the process. However, varying the frequency of magnetic field can trigger different mechanisms, such as natural resonances and exchange resonances, both of which may influence electron spin alignment.¹⁹ Unfortunately, research providing specific parameters for field intensity and frequency settings is lacking. Further investigation into the effects of external magnetic fields on spin



polarization, particularly the impact of magnetic field direction and intensity, could lead to breakthroughs in charge separation and reaction kinetics, offering valuable insights for optimizing electron spins. Additionally, designing electrically, magnetically, or dielectrically sensitive materials as supports for photocatalysts, or utilizing these supports directly as composite catalysts, could enhance response to external fields without compromising the optimal performance of the photocatalyst components.¹⁹

Modulating the metal coordination environment to enhance electron spin polarization has opened new avenues of research in photocatalysis. By meticulously designing the geometric structure and electronic properties of metal ligands, it is possible to improve electron spin polarization in photocatalysts, thereby optimizing the electronic environment of active sites and enabling selective modulation of reaction pathways. Future research should integrate theoretical calculations with experimental investigations to develop a comprehensive understanding of how metal coordination affects electron spin polarization, ultimately guiding photocatalyst optimization. Additionally, crucial to advance synthetic methods that enable precise design of the metal coordination environment.

Integrating chiral materials in photocatalysis has shown promise in inducing spin selectivity, leading to improved charge separation and selective product generation. Future research should concentrate on synthesizing innovative chiral nanostructures and exploring their influence on electron spin. By merging insights from theoretical studies with practical applications, researchers can develop new chiral photocatalysts, such as chiral MOFs, chiral COFs, and chiral halide perovskites.^{159–161} Additionally, a deeper understanding of the mechanisms underlying the CISS effect is essential for fully harnessing its potential. Notably, combining different materials or strategies can create hybrid systems that leverage the advantages of each component, thereby enhancing spin control and photocatalytic efficiency.

2. Developing spin-regulated photocatalysts that fulfill the requirements for practical photocatalytic applications. In practical photocatalytic applications, complex environmental conditions and the presence of multiple components can weaken the spin polarization effect, posing a significant challenge to its practical implementation. To address practical needs, it is imperative to develop stable and efficient spin-regulated photocatalysts for photocatalytic water splitting, CO₂ reduction, pollutant degradation, and N₂ fixation. This will facilitate their integration into industrial and environmental settings.

Certain photocatalysts have the potential for practical applications due to their stability, efficiency, and scalability. For instance, TiO₂ is one of the most widely employed photocatalysts, known for its excellent photostability and non-toxic characteristics. Researchers are actively exploring methods to control its electron spin through techniques such as morphology control, defect engineering, and the integration of chiral materials.^{13,162,163} COFs also offer a promising avenue for development, as their tunable structures facilitate the incorporation of various functional groups and metals, allowing for

precise regulation of electron spin.¹⁶¹ Chiral halide perovskites are potential candidates due to their unique optical properties and ability to induce spin selectivity.¹⁵⁹ Their compositional flexibility imparts the design of photocatalysts that efficiently utilize light energy while maintaining favorable spin characteristics. Overall, future research on spin-regulated photocatalysts should focus on integrating these advanced materials into practical systems, ensuring effective operation under diverse environmental conditions while achieving high efficiency and stability.

3. Innovating synthesis methods to achieve precise control of electron spin. The development of novel synthetic techniques that enable precise tuning of spin-selective sites in photocatalysts will pave the way for materials with optimized spin states. Future research should prioritize integrating molecular and atomic-level spin control mechanisms during synthesis, including the precise introduction of dopants, defects, ligands, and chiral components to promote spin polarization without compromising material stability or scalability. Advanced techniques such as atomic layer deposition and template-assisted methods allow for fine-tuning of morphology, lattice structure, and coordination environment of materials, thereby facilitating precise control over electron spin. Furthermore, combining theoretical modeling with experimental synthesis is crucial for understanding how these modifications impact electron spin. As the field progresses, developing environmentally sustainable and cost-effective synthesis methods will be key to translating these advances into practical applications, driving the next generation of efficient and spin-regulated photocatalysts.

4. Deepening understanding of mechanisms underlying electron spin control. Acquiring a deeper understanding of the mechanisms governing electron spin control is crucial for advancing the development of efficient spin-regulated photocatalysts. Future research should elucidate these mechanisms through a synergistic approach that combines experimental techniques with theoretical calculations. Advanced techniques include XAS, Mössbauer spectroscopy, EELS, ESR spectroscopy, and *M-T* measurement render the scrutiny of local electronic environments and spin dynamics within photocatalysts. Additionally, TRPL and TAS can be employed to investigate the ultrafast processes related to charge carrier dynamics influenced by electron spin control. Furthermore, the generation and transformation of intermediates at active sites of photocatalysts during reaction can be examined using *in situ* techniques such as *in situ* EPR, FTIR, and Raman spectroscopy, offering insights into spin-affected surface reactions.

Theoretical calculations can be utilized to model how various factors such as doping, defects, and coordination environmental affect electron spin behavior. This approach elucidates the underlying mechanisms from a theoretical perspective while offering supplementary analysis of experimental results. Additionally, these calculations enable the prediction and optimization of electronic structure and spin characteristics of photocatalysts. By integrating these theoretical insights with experimental data, we could deepen our understanding of spin control mechanisms and guide the design of new materials and strategies aimed at optimizing photocatalytic performance.



One of the key challenges in mechanism studies is separating the effects of electron spin control from the inherent impacts of modification strategies on photocatalytic performance. Accurately making this distinction is critical, as any ambiguity could result in misinterpretation of the results and lead to flawed conclusions. When investigating the effects of electron spin induced by different strategies, it is essential to first consider how each strategy influences photocatalysis. For instance, defect-induced electron spin polarization can enhance the separation efficiency of photo-generated carriers, while the defects may also act as reaction sites, working synergistically to improve photocatalytic performance. Furthermore, it is crucial to differentiate whether the enhancement in photocatalytic performance stems from the intrinsic effects of strategies like doping or defect engineering, or from the spin polarization induced by these strategies. A combination of experimental and theoretical approaches could clarify this distinction. Comparative experiments using non-spin-active materials or adjusting the strength of spin polarization (e.g., through application of magnetic field) could isolate the spin-related contributions. Advanced characterization techniques, such as EPR and spin-resolved photoelectron spectroscopy (PES), could directly probe spin polarization. Coupled with time-resolved spectroscopy and surface reaction kinetics measurements, these techniques enable the assessment of how spin polarization influences carrier dynamics and reaction pathways. Additionally, theoretical calculations and dynamic simulations could provide insights into the independent effects of spin polarization and structural or electronic modifications.

However, the characterization techniques and theoretical calculations for evaluating electron spin behavior are still in development, posing challenges for research in this area. Future efforts should emphasize interdisciplinary collaboration, integrating expertise from chemistry, physics, materials science, and engineering. This collaborative approach will facilitate the design and optimization of next-generation spin-regulated photocatalysts.

In summary, both opportunities and challenges coexist in the field of spin-regulated photocatalysis. Researchers are encouraged to undertake persistent efforts to advance electron spin control strategies aimed to optimize photocatalysis. This involves transitioning fundamental research into practical applications, including photocatalytic water splitting, CO₂ reduction, pollutant degradation, and N₂ fixation. We hope this review provides valuable insights and inspiration for those interested in leveraging electron spin control to enhance photocatalytic activity.

Data availability

This review does not include any original research results, software, or code, nor were any new data generated or analyzed. Additional information about the review is available upon request from the authors.

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

Z. Lin acknowledges the financial support by the Ministry of Education (MOE), Singapore, under its Academic Research Fund Tier 2 program (T2EP50123-0032) and Agency for Science, Technology and Research (A*STAR) under its RIE2025 MTC IRG program (M23M6c0106). J. Fang acknowledges the financial support by the National Natural Science Foundation of China (22325607 and 22176228). Y. Liu acknowledges the financial support by the National Natural Science Foundation of China (52173207), the Science Fund for Distinguished Young Scholars of Hunan Province of China (2023JJ10040), and the Science and Technology Innovation Program of Hunan Province (2022RC1077). S. He acknowledges the financial support by the China Scholarship Council.

References

- 1 N. S. Lewis, *Science*, 2016, **351**, aad1920.
- 2 X. Tao, Y. Zhao, S. Wang, C. Li and R. Li, *Chem. Soc. Rev.*, 2022, **51**, 3561–3608.
- 3 B. Dai, G. M. Biesold, M. Zhang, H. Zou, Y. Ding, Z. L. Wang and Z. Lin, *Chem. Soc. Rev.*, 2021, **50**, 13646–13691.
- 4 B. Samanta, Á. Morales-García, F. Illas, N. Goga, J. A. Anta, S. Calero, A. Bieberle-Hütter, F. Libisch, A. B. Muñoz-García and M. Pavone, *Chem. Soc. Rev.*, 2022, **51**, 3794–3818.
- 5 S. Fang, M. Rahaman, J. Bharti, E. Reisner, M. Robert, G. A. Ozin and Y. H. Hu, *Nat. Rev. Methods Primers*, 2023, **3**, 61.
- 6 P. Zhou, I. A. Navid, Y. Ma, Y. Xiao, P. Wang, Z. Ye, B. Zhou, K. Sun and Z. Mi, *Nature*, 2023, **613**, 66–70.
- 7 T. Banerjee, F. Podjaski, J. Kröger, B. P. Biswal and B. V. Lotsch, *Nat. Rev. Mater.*, 2021, **6**, 168–190.
- 8 X. Chen, L. Liu, P. Y. Yu and S. S. Mao, *Science*, 2011, **331**, 746–750.
- 9 X. Li, J. Yu, M. Jaroniec and X. Chen, *Chem. Rev.*, 2019, **119**, 3962–4179.
- 10 Z. Zhao, T. Zhang, S. Yue, P. Wang, Y. Bao and S. Zhan, *ChemPhysChem*, 2024, **25**, e202300726.
- 11 T. Wu, X. Ren, Y. Sun, S. Sun, G. Xian, G. G. Scherer, A. C. Fisher, D. Mandler, J. W. Ager and A. Grimaud, *Nat. Commun.*, 2021, **12**, 3634.
- 12 M. Ai, L. Pan, C. Shi, Z.-F. Huang, X. Zhang, W. Mi and J.-J. Zou, *Nat. Commun.*, 2023, **14**, 4562.
- 13 L. Pan, M. Ai, C. Huang, L. Yin, X. Liu, R. Zhang, S. Wang, Z. Jiang, X. Zhang and J.-J. Zou, *Nat. Commun.*, 2020, **11**, 418.
- 14 Y. Zhang, Q. Wu, J. Z. Y. Seow, Y. Jia, X. Ren and Z. J. Xu, *Chem. Soc. Rev.*, 2024, **53**, 8123–8136.
- 15 Z. Zhang, P. Ma, L. Luo, X. Ding, S. Zhou and J. Zeng, *Angew. Chem., Int. Ed.*, 2023, **62**, e202216837.
- 16 R. Naaman, Y. Paltiel and D. H. Waldeck, *Nat. Rev. Chem.*, 2019, **3**, 250–260.
- 17 A. Manchon, H. C. Koo, J. Nitta, S. M. Frolov and R. A. Duine, *Nat. Mater.*, 2015, **14**, 871–882.



18 B. D. Ravetz, N. E. Tay, C. L. Joe, M. Sezen-Edmonds, M. A. Schmidt, Y. Tan, J. M. Janey, M. D. Eastgate and T. Rovis, *ACS Cent. Sci.*, 2020, **6**, 2053–2059.

19 Z. Wang, Y. Li, C. Wu and S. C. E. Tsang, *Joule*, 2022, **6**, 1798–1825.

20 C.-C. Lin, T.-R. Liu, S.-R. Lin, K. M. Boopathi, C.-H. Chiang, W.-Y. Tzeng, W.-H. C. Chien, H.-S. Hsu, C.-W. Luo and H.-Y. Tsai, *J. Am. Chem. Soc.*, 2022, **144**, 15718–15726.

21 Y. Li, Z. Wang, Y. Wang, A. Kovács, C. Foo, R. E. Dunin-Borkowski, Y. Lu, R. A. Taylor, C. Wu and S. C. E. Tsang, *Energy Environ. Sci.*, 2022, **15**, 265–277.

22 D. Zhang, P. Wang, J. Wang, Y. Li, Y. Xia and S. Zhan, *Proc. Natl. Acad. Sci. U. S. A.*, 2021, **118**, e2114729118.

23 D. Zhang, Y. Li, P. Wang, J. Qu, S. Zhan and Y. Li, *Angew. Chem., Int. Ed.*, 2023, **135**, e202303807.

24 S. Sun, G. Shen, J. Jiang, W. Mi, X. Liu, L. Pan, X. Zhang and J. J. Zou, *Adv. Energy Mater.*, 2019, **9**, 1901505.

25 A. Vadakkayil, C. Clever, K. N. Kunzler, S. Tan, B. P. Bloom and D. H. Waldeck, *Nat. Commun.*, 2023, **14**, 1067.

26 X. Ren, T. Wu, Y. Sun, Y. Li, G. Xian, X. Liu, C. Shen, J. Gracia, H.-J. Gao and H. Yang, *Nat. Commun.*, 2021, **12**, 2608.

27 W. Mtangi, F. Tassinari, K. Vankayala, A. Vargas Jentzsch, B. Adelizzi, A. R. Palmans, C. Fontanesi, E. Meijer and R. Naaman, *J. Am. Chem. Soc.*, 2017, **139**, 2794–2798.

28 T. Sun, Z. Tang, W. Zang, Z. Li, J. Li, Z. Li, L. Cao, J. S. Dominic Rodriguez, C. O. M. Mariano and H. Xu, *Nat. Nanotechnol.*, 2023, **18**, 763–771.

29 H. Im, S. Ma, H. Lee, J. Park, Y. S. Park, J. Yun, J. Lee, S. Moon and J. Moon, *Energy Environ. Sci.*, 2023, **16**, 1187–1199.

30 X. Li, J. Yu and M. Jaroniec, *Chem. Soc. Rev.*, 2016, **45**, 2603–2636.

31 Z. Shi, X. Zhang, X. Lin, G. Liu, C. Ling, S. Xi, B. Chen, Y. Ge, C. Tan and Z. Lai, *Nature*, 2023, **621**, 300–305.

32 V. I. Parvulescu, F. Epron, H. Garcia and P. Granger, *Chem. Rev.*, 2021, **122**, 2981–3121.

33 Q. Guo, C. Zhou, Z. Ma and X. Yang, *Adv. Mater.*, 2019, **31**, 1901997.

34 S. Cao, J. Low, J. Yu and M. Jaroniec, *Adv. Mater.*, 2015, **27**, 2150–2176.

35 Q. Li, X. Li, S. Wageh, A. A. Al-Ghamdi and J. Yu, *Adv. Energy Mater.*, 2015, **5**, 1500010.

36 H. Kisch, *Angew. Chem., Int. Ed.*, 2010, **122**, 9782–9783.

37 Y. Chen, L. Soler, C. Cazorla, J. Oliveras, N. G. Bastús, V. F. Puntes and J. Llorca, *Nat. Commun.*, 2023, **14**, 6165.

38 Y. Bai, C. Li, L. Liu, Y. Yamaguchi, M. Bahri, H. Yang, A. Gardner, M. A. Zwijnenburg, N. D. Browning and A. J. Cowan, *Angew. Chem., Int. Ed.*, 2022, **134**, e202201299.

39 B. Liu, X. Wang, Y. Zhang, L. Xu, T. Wang, X. Xiao, S. Wang, L. Wang and W. Huang, *Angew. Chem., Int. Ed.*, 2023, **135**, e202217346.

40 J. Albero, Y. Peng and H. García, *ACS Catal.*, 2020, **10**, 5734–5749.

41 S. He, M. Shen, E. Wu, R. Yin, M. Zhu and L. Zeng, *Environ. Sci. Ecotechnol.*, 2022, **9**, 100141.

42 Z. Wang, N. Hu, L. Wang, H. Zhao and G. Zhao, *Angew. Chem., Int. Ed.*, 2024, **63**, e202407628.

43 S. He, C. Zhai, M. Fujitsuka, S. Kim, M. Zhu, R. Yin, L. Zeng and T. Majima, *Appl. Catal., B*, 2021, **281**, 119479.

44 S. He, Y. Chen, X. Li, L. Zeng and M. Zhu, *ACS ES&T Eng.*, 2022, **2**, 527–546.

45 S. He, L. Ling, Y. Wu, S. Yang, Z. Hua, K. Tang, M. Wang, M. Zhu and J. Fang, *Environ. Sci. Technol.*, 2023, **57**, 9055–9063.

46 J. Qu, D. Zhang, Y. Li, P. Wang, Y. Mao, T. Zhang, S. Zhan and Y. Li, *Appl. Catal., B*, 2024, **340**, 123246.

47 S. He, R. Yin, Y. Chen, T. Lai, W. Guo, L. Zeng and M. Zhu, *Chem. Eng. J.*, 2021, **423**, 130172.

48 L. Wang, W. Wu, K. Liang and X. Yu, *Energy Fuels*, 2022, **36**, 11278–11291.

49 B.-H. Wang, B. Hu, G.-H. Chen, X. Wang, S. Tian, Y. Li, X.-S. Hu, H. Wang, C.-T. Au and L.-L. Jiang, *Chem. Catal.*, 2024, **4**, 101083.

50 Z. Zhao, D. Wang, R. Gao, G. Wen, M. Feng, G. Song, J. Zhu, D. Luo, H. Tan and X. Ge, *Angew. Chem., Int. Ed.*, 2021, **60**, 11910–11918.

51 P. Wang, J. Wang, T. Ming, X. Wang, H. Yu, J. Yu, Y. Wang and M. Lei, *ACS Appl. Mater. Interfaces*, 2013, **5**, 2924–2929.

52 Y. Lei, C. Yang, J. Hou, F. Wang, S. Min, X. Ma, Z. Jin, J. Xu, G. Lu and K.-W. Huang, *Appl. Catal., B*, 2017, **216**, 59–69.

53 M. Zhu, S. Kim, L. Mao, M. Fujitsuka, J. Zhang, X. Wang and T. Majima, *J. Am. Chem. Soc.*, 2017, **139**, 13234–13242.

54 W. Yang, L. Zhang, Y. Hu, Y. Zhong, H. B. Wu and X. W. Lou, *Angew. Chem., Int. Ed.*, 2012, **51**, 11501–11504.

55 H. Yang, Y. Huang, B. Luo, Z. Xie, D. Li, D. Xu, Y. Lei and W. Shi, *Chem. Commun.*, 2024, **60**, 1035–1038.

56 R. Hao, G. Wang, H. Tang, L. Sun, C. Xu and D. Han, *Appl. Catal., B*, 2016, **187**, 47–58.

57 W. Wang, J. Fang, S. Shao, M. Lai and C. Lu, *Appl. Catal., B*, 2017, **217**, 57–64.

58 F. Lin, D. Wang, Z. Jiang, Y. Ma, J. Li, R. Li and C. Li, *Energy Environ. Sci.*, 2012, **5**, 6400–6406.

59 J. Xiong, J. Di, J. Xia, W. Zhu and H. Li, *Adv. Funct. Mater.*, 2018, **28**, 1801983.

60 J. Ran, B. Zhu and S. Z. Qiao, *Angew. Chem., Int. Ed.*, 2017, **56**, 10373–10377.

61 S. Bai, N. Zhang, C. Gao and Y. Xiong, *Nano Energy*, 2018, **53**, 296–336.

62 J. Hao, X. Xu, H. Fei, L. Li and B. Yan, *Adv. Mater.*, 2018, **30**, 1705634.

63 I. Liu, L. A. Lawton and P. K. Robertson, *Environ. Sci. Technol.*, 2003, **37**, 3214–3219.

64 W. Pauli, *Naturwissenschaften*, 1924, **12**, 741.

65 A. Pais, *Phys. Today*, 1989, **42**, 34–40.

66 F. Hund, *Z. Angew. Phys.*, 1927, **40**, 742–764.

67 P. A. M. Dirac, *Proc. R. Soc. London, Ser. A*, 1928, **117**, 610–624.

68 Y.-N. Gong, W. Zhong, Y. Li, Y. Qiu, L. Zheng, J. Jiang and H.-L. Jiang, *J. Am. Chem. Soc.*, 2020, **142**, 16723–16731.

69 J. Suntivich, H. A. Gasteiger, N. Yabuuchi, H. Nakanishi, J. B. Goodenough and Y. Shao-Horn, *Nat. Chem.*, 2011, **3**, 546–550.



70 W. H. Lee, M. H. Han, Y.-J. Ko, B. K. Min, K. H. Chae and H.-S. Oh, *Nat. Commun.*, 2022, **13**, 605.

71 Y. Tong, Y. Guo, P. Chen, H. Liu, M. Zhang, L. Zhang, W. Yan, W. Chu, C. Wu and Y. Xie, *Chem.*, 2017, **3**, 812–821.

72 J. Li, M. T. Sougrati, A. Zitolo, J. M. Ablett, I. C. Oğuz, T. Mineva, I. Matanovic, P. Atanassov, Y. Huang and I. Zenyuk, *Nat. Catal.*, 2021, **4**, 10–19.

73 Y. Liu, X. Liu, Z. Lv, R. Liu, L. Li, J. Wang, W. Yang, X. Jiang, X. Feng and B. Wang, *Angew. Chem., Int. Ed.*, 2022, **134**, e202117617.

74 M. T. Sougrati, V. Goellner, A. K. Schuppert, L. Stievano and F. Jaouen, *Catal. Today*, 2016, **262**, 110–120.

75 C. Ma, N. Lin, Z. Wang, S. Zhou, H. Yu, J. Lu and H. Huang, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2019, **99**, 115401.

76 Q. Lan, X. Zhang, X. Shen, H. Yang, H. Zhang, X. Guan, W. Wang, Y. Yao, Y. Wang and Y. Peng, *Phys. Rev. Mater.*, 2017, **1**, 024403.

77 S. A. Bonke, T. Risse, A. Schnegg and A. Brückner, *Nat. Rev. Methods Primers*, 2021, **1**, 33.

78 C. Wu, X. Wang, Y. Tang, H. Zhong, X. Zhang, A. Zou, J. Zhu, C. Diao, S. Xi and J. Xue, *Angew. Chem., Int. Ed.*, 2023, **135**, e202218599.

79 S. Zhou, X. Miao, X. Zhao, C. Ma, Y. Qiu, Z. Hu, J. Zhao, L. Shi and J. Zeng, *Nat. Commun.*, 2016, **7**, 11510.

80 Z.-D. He, R. Tesch, M. J. Eslamibidgoli, M. H. Eikerling and P. M. Kowalski, *Nat. Commun.*, 2023, **14**, 3498.

81 X. Zhang, H. Zhong, Q. Zhang, Q. Zhang, C. Wu, J. Yu, Y. Ma, H. An, H. Wang and Y. Zou, *Nat. Commun.*, 2024, **15**, 1383.

82 A. Moroz and C. Barnes, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1999, **60**, 14272.

83 R. Schaffer, E. K.-H. Lee, B.-J. Yang and Y. B. Kim, *Rep. Prog. Phys.*, 2016, **79**, 094504.

84 F. Schwabl, *Quantum Mechanics*, Springer, 2007.

85 S. Jiao, X. Fu and H. Huang, *Adv. Funct. Mater.*, 2022, **32**, 2107651.

86 J. Suntivich, K. J. May, H. A. Gasteiger, J. B. Goodenough and Y. Shao-Horn, *Science*, 2011, **334**, 1383–1385.

87 A. Grimaud, K. J. May, C. E. Carlton, Y.-L. Lee, M. Risch, W. T. Hong, J. Zhou and Y. Shao-Horn, *Nat. Commun.*, 2013, **4**, 2439.

88 G. Yang, J. Zhu, P. Yuan, Y. Hu, G. Qu, B.-A. Lu, X. Xue, H. Yin, W. Cheng and J. Cheng, *Nat. Commun.*, 2021, **12**, 1734.

89 M.-H. Whangbo, E. E. Gordon, H. Xiang, H.-J. Koo and C. Lee, *Acc. Chem. Res.*, 2015, **48**, 3080–3087.

90 Y. Sun, S. Sun, H. Yang, S. Xi, J. Gracia and Z. J. Xu, *Adv. Mater.*, 2020, **32**, 2003297.

91 I. Bauer and H.-J. Knölker, *Chem. Rev.*, 2015, **115**, 3170–3387.

92 A. L. Buchachenko and V. L. Berdinsky, *Chem. Rev.*, 2002, **102**, 603–612.

93 Y. Li, D. Zhang, P. Wang, J. Qu and S. Zhan, *Proc. Natl. Acad. Sci. U. S. A.*, 2024, **121**, e2407012121.

94 C. Deng, S. Xie, Y. Li, Y. Zhao, P. Zhou, H. Sheng, H. Ji, C. Chen and J. Zhao, *J. Phys. Chem. A*, 2023, **127**, 2787–2794.

95 Y. Liu, Q. Deng, Z. Yao, T. Liang, S. Zhang, T. Zhu, C. Xing, J. Pan, Z. Yu and K. Liang, *J. Colloid Interface Sci.*, 2024, **664**, 500–510.

96 W. Li, Y. Zhang, W. Ran, Y. Wang, F. Tian, F. Zhang, M. Xu, D. Zhang, N. Li and T. Yan, *Appl. Catal., B*, 2024, **351**, 123978.

97 Y. Mao, B. Yu, P. Wang, S. Yue and S. Zhan, *Nat. Commun.*, 2024, **15**, 6364.

98 C. Cheng, W.-H. Fang, R. Long and O. V. Prezhdo, *JACS Au*, 2021, **1**, 550–559.

99 C. Zhu, L. Lu, Q. Fang, S. Song, B. Chen and Y. Shen, *Adv. Funct. Mater.*, 2023, **33**, 2210905.

100 M. Li, S. Wu, D. Liu, Z. Ye, L. Wang, M. Kan, Z. Ye, M. Khan and J. Zhang, *J. Am. Chem. Soc.*, 2024, **146**, 15538–15548.

101 R. Ramprasath, V. Manikandan, S. Aldawood, S. Sudha, S. Cholan, N. Kannadasan, S. Sampath and B. Gokul, *Environ. Res.*, 2022, **214**, 113866.

102 X. Zhan, J. Liu, Y. Zhao, Y. Sun, R. Gao, H. Wang and H. Shi, *Sep. Purif. Technol.*, 2022, **302**, 122146.

103 Y. Wang, Y. Xie, S. Yu, K. Yang, Y. Shao, L. Zou, B. Zhao, Z. Wang, Y. Ling and Y. Chen, *Appl. Catal., B*, 2023, **327**, 122420.

104 J. Wan, L. Liu, Y. Wu, J. Song, J. Liu, R. Song, J. Low, X. Chen, J. Wang and F. Fu, *Adv. Funct. Mater.*, 2022, **32**, 2203252.

105 C. Zhu, Q. Fang, R. Liu, W. Dong, S. Song and Y. Shen, *Environ. Sci. Technol.*, 2022, **56**, 6699–6709.

106 J. Luo, C. Feng, C. Fan, L. Tang, Y. Liu, Z. Gong, T. Wu, X. Zhen, H. Feng and M. Yan, *J. Catal.*, 2022, **413**, 1132–1145.

107 X. Zhang and G. Lu, *Carbon*, 2016, **108**, 215–224.

108 Y. Wang, W. Xu, Y. Zhang, Y. Wu, Z. Wang, L. Fu, F. Bai, B. Zhou, T. Wang and L. Cheng, *Nano Energy*, 2021, **83**, 105783.

109 Q. Wang, X. Deng, H. Pen, F. Liu, M. Song, P. Chen and S.-F. Yin, *Nano Res.*, 2023, **16**, 4225–4232.

110 Y. Lin, Z. Jiang, C. Zhu, X. Hu, X. Zhang, H. Zhu, J. Fan and S. H. Lin, *J. Mater. Chem. A*, 2013, **1**, 4516–4524.

111 C. Zhu, L. Lu, J. Xu, S. Song, Q. Fang, R. Liu, Y. Shen, J. Zhao, W. Dong and Y. Shen, *Chem. Eng. J.*, 2023, **451**, 138537.

112 X. Yang, F. Li, W. Liu, L. Chen, J. Qi, W. Sun, F. Pan, T. Duan and F. Sun, *Appl. Catal., B*, 2023, **324**, 122202.

113 J. He, L. Hu, C. Shao, S. Jiang, C. Sun and S. Song, *ACS Nano*, 2021, **15**, 18006–18013.

114 W. Wu and L. Yang, *J. Catal.*, 2024, **429**, 115283.

115 Z. Jin, J. Zhang, J. Qiu, Y. Hu, T. Di and T. Wang, *J. Colloid Interface Sci.*, 2023, **652**, 122–131.

116 Z. Qi, J. Chen, Q. Li, N. Wang, S. A. Carabineiro and K. Lv, *Small*, 2023, **19**, 2303318.

117 J. Xu, L. Lu, C. Zhu, Q. Fang, R. Liu, D. Wang, Z. He, S. Song and Y. Shen, *J. Colloid Interface Sci.*, 2023, **630**, 430–442.

118 J. Li, Q. Pei, R. Wang, Y. Zhou, Z. Zhang, Q. Cao, D. Wang, W. Mi and Y. Du, *ACS Nano*, 2018, **12**, 3351–3359.

119 K. Sun, Y. Huang, Q. Wang, W. Zhao, X. Zheng, J. Jiang and H.-L. Jiang, *J. Am. Chem. Soc.*, 2024, **146**, 3241–3249.



120 G.-Z. Huang, Y.-S. Xia, F. Yang, W.-J. Long, J.-J. Liu, J.-P. Liao, M. Zhang, J. Liu and Y.-Q. Lan, *J. Am. Chem. Soc.*, 2023, **145**, 26863–26870.

121 T. H. Kim, K. Cho, S. H. Lee, J. H. Kang, H. B. Park, J. Park and Y.-H. Kim, *Chem. Eng. J.*, 2024, **492**, 152095.

122 Y. Sun, J. Suriyaprakash, L. Shan, H. Xu, J. Zhang, G. Chen, Y. Zhang, H. Wu, X. Li and L. Dong, *Appl. Catal., B*, 2024, **355**, 124209.

123 J. Li, D. Hu and Q. Chen, *Ceram. Int.*, 2024, **50**, 5293–5310.

124 G. Li, X. Sun, P. Chen, M. Song, T. Zhao, F. Liu and S.-F. Yin, *Nano Res.*, 2023, **16**, 8845–8852.

125 W. Gao, X. Zhao, T. Zhang, X. Yu, Y. Ma, E. C. dos Santos, J. White, H. Liu and Y. Sang, *Nano Energy*, 2023, **110**, 108381.

126 M. Li, S. Wu, D. Liu, Z. Ye, C. He, J. Wang, X. Gu, Z. Zhang, H. Li and J. Zhang, *ACS Catal.*, 2024, **14**, 14098–14109.

127 C. Xing, Z. Yu, N. A. Al-Dhabi, Z. Yao, T. Zhu, Y. Liu, J. Pan, W. Tang and Y. Hou, *Sep. Purif. Technol.*, 2025, **354**, 129030.

128 C.-H. Chiang, C.-C. Lin, Y.-C. Lin, C.-Y. Huang, C.-H. Lin, Y.-J. Chen, T.-R. Ko, H.-L. Wu, W.-Y. Tzeng and S.-Z. Ho, *J. Am. Chem. Soc.*, 2024, **146**, 23278–23288.

129 F. Wu, X. Zhang, L. Wang, G. Li, J. Huang, A. Song, A. Meng and Z. Li, *Small*, 2024, **20**, 2309439.

130 S. Wei, J. Zhang, L. Zhang, Y. Wang, H. Sun, X. Hua, Z. Guo and D. Dong, *Appl. Catal., B*, 2024, **358**, 124406.

131 G. Yao, S. Yang, S. Jiang, C. Sun and S. Song, *Appl. Catal., B*, 2022, **315**, 121569.

132 Y. Wang, W. Xu, Y. Zhang, C. Zeng, W. Zhang, L. Fu, M. Sun, Y. Wu, J. Hao and W. Zhong, *Energy Environ. Mater.*, 2023, **6**, e12390.

133 F. Li, T. Huang, F. Sun, L. Chen, P. Li, F. Shao, X. Yang and W. Liu, *Appl. Catal., B*, 2022, **317**, 121725.

134 Z. He, K. Lin, N. H. Wong, J. Sunarso, Y. Xia, X. Fu, B. Tang, Z. Huang, Y. Wang and H. Yang, *Nano Energy*, 2024, **124**, 109483.

135 I. Zanella, S. Guerini, S. Fagan, J. Mendes Filho and A. Souza Filho, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2008, **77**, 073404.

136 A. Zhang, Y. Liang, H. Zhang, Z. Geng and J. Zeng, *Chem. Soc. Rev.*, 2021, **50**, 9817–9844.

137 V.-H. Do and J.-M. Lee, *ACS Nano*, 2022, **16**, 17847–17890.

138 A. Srivastava, M. Sidler, A. V. Allain, D. S. Lembke, A. Kis and A. Imamoğlu, *Nat. Phys.*, 2015, **11**, 141–147.

139 L. Jiao, J. Wang and H.-L. Jiang, *Acc. Chem. Res.*, 2021, **2**, 327–339.

140 K. Chae, N. A. R. C. Mohamad, J. Kim, D.-I. Won, Z. Lin, J. Kim and D. H. Kim, *Chem. Soc. Rev.*, 2024, **53**, 9029–9058.

141 B. P. Bloom, Y. Paltiel, R. Naaman and D. H. Waldeck, *Chem. Rev.*, 2024, **124**, 1950–1991.

142 S. Ma, H. Lee and J. Moon, *Adv. Mater.*, 2024, **36**, 2405685.

143 W. Fu, L. Tan and P.-P. Wang, *ACS Nano*, 2023, **17**, 16326–16347.

144 M. V. Mukhina, V. G. Maslov, A. V. Baranov, A. V. Fedorov, A. O. Orlova, F. Purcell-Milton, J. Govan and Y. K. Gun'ko, *Nano Lett.*, 2015, **15**, 2844–2851.

145 C.-C. Chen, C. Zhu, E. R. White, C.-Y. Chiu, M. Scott, B. Regan, L. D. Marks, Y. Huang and J. Miao, *Nature*, 2013, **496**, 74–77.

146 Z. Dong and Y. Ma, *Nat. Commun.*, 2020, **11**, 1588.

147 J. K. Choi, B. E. Haynie, U. Tohgha, L. Pap, K. W. Elliott, B. M. Leonard, S. V. Dzyuba, K. Varga, J. Kubelka and M. Balaz, *ACS Nano*, 2016, **10**, 3809–3815.

148 J. Cheng, J. Hao, H. Liu, J. Li, J. Li, X. Zhu, X. Lin, K. Wang and T. He, *ACS Nano*, 2018, **12**, 5341–5350.

149 Y. Li, J. Cheng, J. Li, X. Zhu, T. He, R. Chen and Z. Tang, *Angew. Chem., Int. Ed.*, 2018, **130**, 10393–10397.

150 V. A. Kuznetsova, E. Mates-Torres, N. Prochukhan, M. Marcastel, F. Purcell-Milton, J. O'Brien, A. K. Visheratina, M. Martinez-Carmona, Y. Gromova and M. Garcia-Melchor, *ACS Nano*, 2019, **13**, 13560–13572.

151 H.-E. Lee, H.-Y. Ahn, J. Mun, Y. Y. Lee, M. Kim, N. H. Cho, K. Chang, W. S. Kim, J. Rho and K. T. Nam, *Nature*, 2018, **556**, 360–365.

152 L. Ohnoutek, J.-Y. Kim, J. Lu, B. J. Olohan, D. M. Răsădean, G. Dan Pantos, N. A. Kotov and V. K. Valev, *Nat. Photonics*, 2022, **16**, 126–133.

153 B. Ni, M. Mychinko, S. Gómez-Graña, J. Morales-Vidal, M. Obelleiro-Liz, W. Heyvaert, D. Vila-Liarte, X. Zhuo, W. Albrecht and G. Zheng, *Adv. Mater.*, 2023, **35**, 2208299.

154 W. Yan, L. Xu, C. Xu, W. Ma, H. Kuang, L. Wang and N. A. Kotov, *J. Am. Chem. Soc.*, 2012, **134**, 15114–15121.

155 X. Lan, X. Lu, C. Shen, Y. Ke, W. Ni and Q. Wang, *J. Am. Chem. Soc.*, 2015, **137**, 457–462.

156 A. Kuzyk, R. Schreiber, H. Zhang, A. O. Govorov, T. Liedl and N. Liu, *Nat. Mater.*, 2014, **13**, 862–866.

157 L. Ma, Y. Liu, C. Han, A. Movsesyan, P. Li, H. Li, P. Tang, Y. Yuan, S. Jiang and W. Ni, *Nano Lett.*, 2022, **22**, 4784–4791.

158 M. Dc, R. Grassi, J.-Y. Chen, M. Jamali, D. Reifsnyder Hickey, D. Zhang, Z. Zhao, H. Li, P. Quarterman and Y. Lv, *Nat. Mater.*, 2018, **17**, 800–807.

159 G. Long, R. Sabatini, M. I. Saidaminov, G. Lakhwani, A. Rasmita, X. Liu, E. H. Sargent and W. Gao, *Nat. Rev. Mater.*, 2020, **5**, 423–439.

160 W. Weng and J. Guo, *J. Am. Chem. Soc.*, 2024, **146**, 13201–13209.

161 X. Han, C. Yuan, B. Hou, L. Liu, H. Li, Y. Liu and Y. Cui, *Chem. Soc. Rev.*, 2020, **49**, 6248–6272.

162 S. Liu, L. Han, Y. Duan, S. Asahina, O. Terasaki, Y. Cao, B. Liu, L. Ma, J. Zhang and S. Che, *Nat. Commun.*, 2012, **3**, 1215.

163 Y. Chang, C. Dong, D. Zhou, A. Li, W. Dong, X.-Z. Cao and G. Wang, *ACS Nano*, 2021, **15**, 14174–14184.

