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# Enhanced NH<sub>3</sub> uptake and selectivity at low pressure in monolithic MOF-808 metal–organic gels incorporating CuCl<sub>2</sub>

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The capture and separation of trace NH<sub>3</sub> from industrial processes or polluted air remains a significant challenge. Herein, we report a monolithic CuCl<sub>2</sub>@G808 metal–organic gel achieving an NH<sub>3</sub> uptake of 2.23 mmol g<sup>-1</sup> at 298 K and 0.002 bar: a 79% enhancement compared to pristine G808. The ideal adsorbed solution theory (IAST) selectivity reaches 2.8 × 10<sup>3</sup> for NH<sub>3</sub>/N<sub>2</sub> and 4.9 × 10<sup>5</sup> for NH<sub>3</sub>/H<sub>2</sub> at 298 K, ranking among the highest reported values. *In situ* FTIR and XPS analyses reveal that the excellent performance mainly originates from two synergistic mechanisms: (i) coordination between NH<sub>3</sub> and Cu<sup>2+</sup> sites, and (ii) hydrogen bonding between NH<sub>3</sub> and Cl<sup>-</sup> sites.

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## Introduction

Ammonia (NH<sub>3</sub>) is an important chemical feedstock, mainly produced by the Haber–Bosch process, and widely used in refrigeration, energy, and fertilizer industries.<sup>1,2</sup> At the same time, ammonia is a highly toxic and corrosive gas, mostly derived from the inadvertent or intentional emission, which causes severe damage to human health and the environment even at low concentrations.<sup>3</sup> Therefore, efficient capture and separation of trace NH<sub>3</sub> from the NH<sub>3</sub> production process or polluted air is of great significance for improving NH<sub>3</sub> production efficiency and reducing energy consumption, and alleviating impacts on the environment and human health.

Metal–organic frameworks (MOFs) adopt structural designability and diversity, abundant active sites, and high porosity, enabling them prospective candidates for real-world NH<sub>3</sub> capture and separation. Several representative MOFs such as LiCl@MIL-53-(OH)<sub>2</sub>-43.4,<sup>4</sup> LiCl@G66-OH-35.7,<sup>5</sup> IL@MIL-101(Cr),<sup>6</sup> Mg<sub>2</sub>(dobpdc),<sup>7</sup> Ni<sub>2</sub>acryl<sub>2</sub>TMA,<sup>8</sup> Cu<sub>2</sub>Cl<sub>2</sub>BBTA,<sup>9</sup> MOF-303(Al),<sup>10</sup> MOF-253(Al)-NiCl<sub>2</sub>-2,<sup>11</sup> MFU-4,<sup>12</sup> Cu(cyhd),<sup>13</sup> Co(NA)<sub>2</sub>,<sup>14</sup> MFM-300(V<sup>IV</sup>),<sup>15</sup> Cu(BDC),<sup>16</sup> UiO-66-Cu<sup>II</sup>,<sup>17</sup> DUT-6-(OH)<sub>2</sub>,<sup>18</sup> [Mn<sub>2</sub>Cl<sub>2</sub>BTDD],<sup>19</sup> Fe-soc-MOF,<sup>20</sup> and MFM-300(Sc)<sup>21</sup> have demonstrated outstanding NH<sub>3</sub> capture performance even at low pressure. However, their connatural limitations such as low NH<sub>3</sub> adsorption capacity and separation ability at extremely low pressure, high production cost, limited stability, and powder state problem preclude their potential application for

NH<sub>3</sub> efficient capture and separation. Therefore, there is an urgent need to fabricate granular high-performance MOFs adsorbents possessing remarkable NH<sub>3</sub> adsorption and separation ability at ultra-low pressure and facile preparation process.

Metal–organic framework gels (MOGs), as a novel self-shaping material, have come to prominent attention owing to its adjustable aperture from micropore to mesopore/macropore, high adsorptive capacity, and easy large-scale preparation.<sup>22</sup> MOGs can be facilely fabricated by regulating reaction conditions such as metal source, reactant concentration, solvent, and temperature.<sup>23,24</sup> Up to now, MOGs have demonstrated excellent capture performance in many fields such as methane storage,<sup>25,26</sup> CO<sub>2</sub> capture and storage,<sup>27,28</sup> volatile organic compounds capture,<sup>29–31</sup> water remediation,<sup>32,33</sup> chemical warfare agents decontamination,<sup>34,35</sup> toxic chemical filtration.<sup>36–38</sup> However, there are few reports on the efficient capture and separation of NH<sub>3</sub> using MOGs, especially under ultra-low pressure.

CuCl<sub>2</sub> possesses an exceptional NH<sub>3</sub> uptake, but its application is greatly limited due to its powder state. Based on the above advantages of MOGs, herein, MOF-808 metal–organic framework gel (labeled as G808) is chosen as the platform due to its excellent stability to NH<sub>3</sub>. Thus, CuCl<sub>2</sub>@G808 composite is prepared in water by a facile impregnation strategy at 80 °C. CuCl<sub>2</sub>@G808 containing 2.55 wt% of CuCl<sub>2</sub> shows excellent low-pressure NH<sub>3</sub> uptake (2.23 mmol g<sup>-1</sup>) at 298 K and 0.002 bar, which displays enhancement of 79% than that of the pristine G808. Notable, highly selective adsorption of trace NH<sub>3</sub> was also obtained at 298 K. Furthermore, molecular-level insights into the adsorption mechanism were elucidated through combined spectroscopic analyses.

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## Result and discussion

### Structure and morphology of CuCl<sub>2</sub>@G808-X composites

G808 is firstly synthesized according to our previous work.<sup>39</sup> Then, CuCl<sub>2</sub>@G808 composite is obtained by a facile impregnation strategy in water solution containing CuCl<sub>2</sub> at 80 °C for 24 h. The content of CuCl<sub>2</sub> loaded on G808 is 2.55% measured by ICP-OES. As observed in Fig. 1a, the PXRD pattern of G808 matches well with the simulated MOF-808 (CCDC: 1002672), as evidenced by the characteristic peaks at  $2\theta = 4.4^\circ$  and  $8.6^\circ$ .<sup>40</sup> Furthermore, the crystallinity of CuCl<sub>2</sub>@G808 decreases compared with that of G808, suggesting that the crystal structure of the composite collapses after loading CuCl<sub>2</sub>. SEM results also further confirm this conclusion (Fig. S1). In addition, no significant additional peaks are observed in the PXRD of the composite, indicating that CuCl<sub>2</sub> is evenly anchored in the nanopores of G808. Meanwhile, the uniform distribution of Cu and Cl elements in CuCl<sub>2</sub>@G808 is verified by energy-dispersive X-ray spectroscopy (Fig. S2). Fig. 1b illustrates the TGA profiles of the samples. The initial mass reduction (303–373 K) is due to residual solvent volatilization, followed by two distinct degradation stages: framework collapse at 473–673 K and organic ligand decomposition at 673–1073 K.<sup>40</sup> Compared with G808, the thermal stability of CuCl<sub>2</sub>@G808 slightly decreases from 743 K to 703 K, indicating that the introduction of CuCl<sub>2</sub> has a certain influence on the thermal stability of the composite. This result is in coincidence with that of XRD. Comparative FT-

IR analyses reveal negligible spectral differences between pristine G808 and CuCl<sub>2</sub>@G808 (Fig. 1c). Compared with G808, the peak of CuCl<sub>2</sub>@G808 stemmed from the carboxylate ( $-\text{COO}^-$ ) group undergoes some degrees of red shift from  $1572\text{ cm}^{-1}$  to  $1562\text{ cm}^{-1}$ , possibly through partial charge transfer from  $-\text{COO}^-$  groups to  $\text{Cu}^{2+}$ .<sup>41</sup> Like G808, the CuCl<sub>2</sub>@G808 composite also display characteristic IV isotherms for N<sub>2</sub> adsorption (Fig. 1d), indicating the existence of microporous and mesoporous features. This result is also validated by the pore size distribution (Fig. S3 and S4). Compared to G808, pore volume, surface area, and pore width of CuCl<sub>2</sub>@G808 composite dramatically decrease due to the occupation of the pore space by CuCl<sub>2</sub> (Table S1).

### NH<sub>3</sub> capture and separation

To study capture performance, NH<sub>3</sub> adsorption isotherms (Fig. 2) of G808 and CuCl<sub>2</sub>@G808 composite are collected at 298 K and 1 bar. The isotherms show obvious hysteresis loops, suggesting the existing of strong interaction between NH<sub>3</sub> and active sites. Compared with G808 ( $9.6\text{ mmol g}^{-1}$ ), the NH<sub>3</sub> capture capacity of the CuCl<sub>2</sub>@G808 ( $8.1\text{ mmol g}^{-1}$ ) slightly decreases due to the decrease of porosity at 298 K and 1 bar. However, the CuCl<sub>2</sub>@G808 manifests a significantly enhanced uptake at low pressure. For example, the NH<sub>3</sub> adsorption capacity of CuCl<sub>2</sub>@G808 can reach approximately  $5.44\text{ mmol g}^{-1}$  at 0.1 bar,  $3.65\text{ mmol g}^{-1}$  at 0.01 bar, and  $2.23\text{ mmol g}^{-1}$  even at 0.002 bar (Fig. 2a), Compared to pristine G808, these

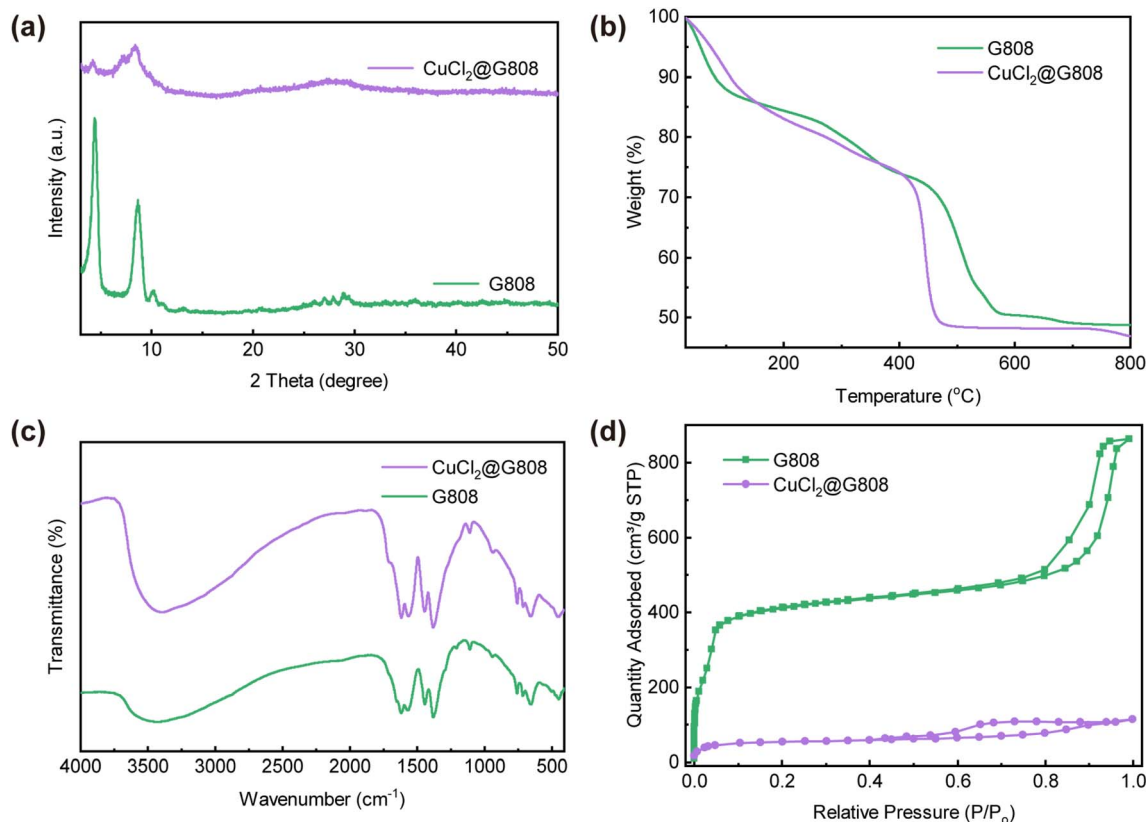


Fig. 1 (a) PXRD patterns, (b) TGA curves, (c) FT-IR spectra, and (d) N<sub>2</sub> adsorption–desorption isotherms of CuCl<sub>2</sub>@G808.



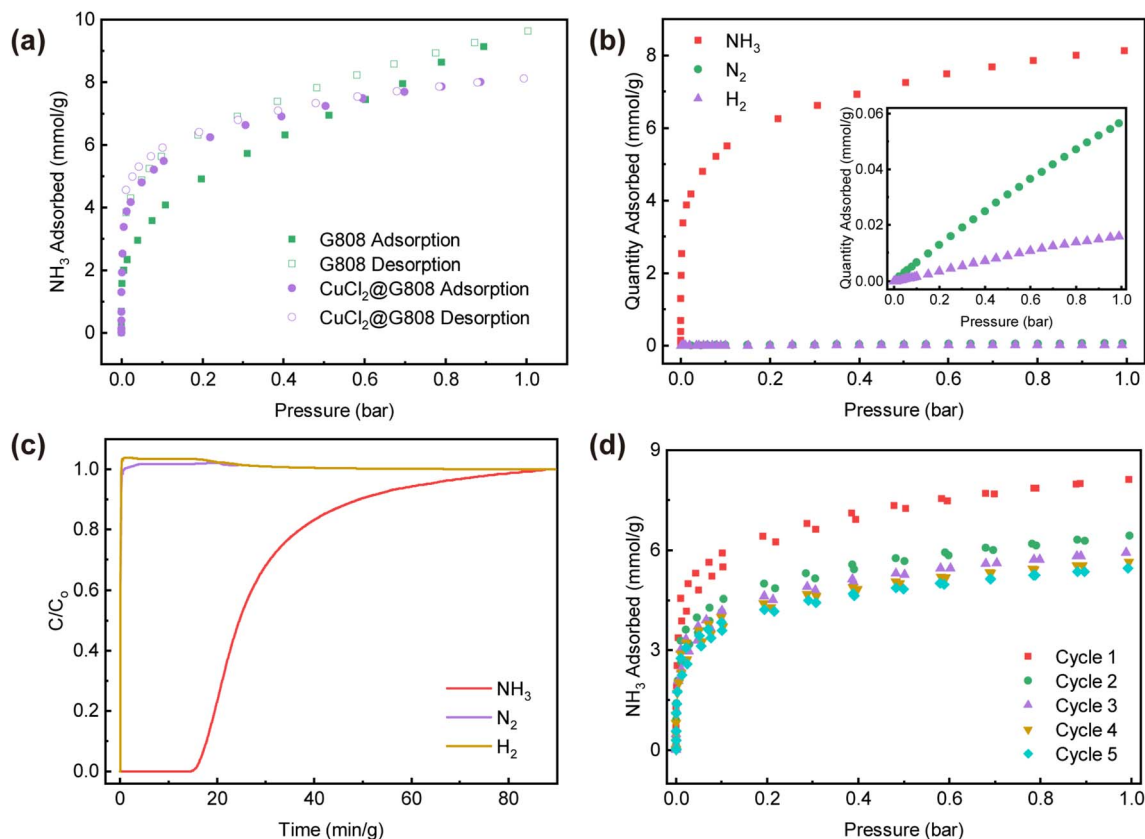


Fig. 2 (a) Adsorption–desorption isotherms of  $\text{NH}_3$ , (b) adsorption isotherms of  $\text{CuCl}_2@G808$  for  $\text{NH}_3$ ,  $\text{N}_2$ , and  $\text{H}_2$ , (c) breakthrough curves of  $\text{CuCl}_2@G808$  for  $\text{NH}_3/\text{N}_2/\text{H}_2$  mixtures, (d)  $\text{NH}_3$  adsorption–desorption isotherms for different cycles of  $\text{CuCl}_2@G808$  at 298 K.

values represent enhancement of 37%, 74%, and 79%, respectively. These results are comparable to the best-behaving values reported recently (Table S2), and indicate that our composite is conducive to the capture of low concentration  $\text{NH}_3$ . To validate this,  $\text{N}_2$  and  $\text{H}_2$  isotherms of  $\text{CuCl}_2@G808$  are measured at 273 K and 298 K (Fig. 2b, S5 and S6). Notably, the adsorption capacities of  $\text{N}_2$  ( $0.056 \text{ mmol g}^{-1}$ ) and  $\text{H}_2$  ( $0.016 \text{ mmol g}^{-1}$ ) on  $\text{CuCl}_2@G808$  at 298 K and 1 bar are significantly lower than that of  $\text{NH}_3$  ( $8.1 \text{ mmol g}^{-1}$ ). This may be attributed to their weak affinity with the composite originated from low  $Q_{\text{st}}$  values (Fig. S7 and S8). The IAST-predicted selectivity values of  $\text{CuCl}_2@G808$  reach  $2.8 \times 10^3$  for  $\text{NH}_3/\text{N}_2$  and  $4.9 \times 10^5$  for  $\text{NH}_3/\text{H}_2$  at 298 K, respectively (Fig. S9 and 10), surpassing most reported MOFs.<sup>11,42</sup> Furthermore, granular  $\text{CuCl}_2@G808$  exhibits excellent self-shaping ability, facile synthesis, and low production cost, demonstrating its competitive and practical potential in low concentration  $\text{NH}_3$  capture and separation.

Considering the valuable  $\text{NH}_3$  industry synthesis, breakthrough experiments are adopted to validate the practical separation capability of  $\text{CuCl}_2@G808$  for  $\text{NH}_3$  using actual  $\text{NH}_3/\text{N}_2/\text{H}_2$  mixtures (3%  $\text{NH}_3$  in 25%  $\text{N}_2$  and 72%  $\text{H}_2$ ) (Fig. 2c). It is clearly observed that  $\text{N}_2$  and  $\text{H}_2$  are immediately escaped because of their low adsorption capacity, followed by  $\text{NH}_3$  after  $16.8 \text{ min g}^{-1}$  accompanied by uptake of  $3.9 \text{ mmol g}^{-1}$ . Thus,  $\text{CuCl}_2@G808$  can be considered as a superior adsorbent for selective separation of  $\text{NH}_3/\text{N}_2/\text{H}_2$  at low  $\text{NH}_3$  concentration.

Five recycling experiment of  $\text{CuCl}_2@G808$  is also carried out to study its practical performance (Fig. 2d and S11).  $\text{NH}_3$  adsorption capacity of  $\text{CuCl}_2@G808$  can be only maintained up to  $5.4 \text{ mmol g}^{-1}$  (67.1%) after five cycles at 298 K and 1 bar due to its partial structural collapse (Fig. S12 and S13). Meanwhile, about 50.7% of  $\text{NH}_3$  is difficult to desorb due to the strong interaction between  $\text{NH}_3$  and  $\text{CuCl}_2@G808$ .

### Mechanism of $\text{NH}_3$ adsorption

To elucidate adsorption mechanism between  $\text{CuCl}_2@G808$  and  $\text{NH}_3$ , a combination of spectroscopic techniques-including *in situ* FTIR and XPS spectra are employed (Fig. S14). After adsorption of  $\text{NH}_3$ , a new IR peak attributed to N–H deformation vibration emerges at  $685 \text{ cm}^{-1}$  (Fig. 3a), suggesting potential coordination between  $\text{NH}_3$  and  $\text{Cu}^{2+}$ .<sup>43</sup> Besides, the C=O stretching vibration peak shifts from  $1566 \text{ cm}^{-1}$  to  $1574 \text{ cm}^{-1}$  (Fig. 3a) and the weak phenyl C–H peak at  $3088 \text{ cm}^{-1}$  disappear (Fig. 3b) after  $\text{NH}_3$  adsorption, possibly due to hydrogen bonding between  $\text{NH}_3$  and these groups.

The XPS survey spectrum of  $\text{CuCl}_2@G808$  over the range of 0–1300 eV is presented in Fig. S15. Characteristic peaks corresponding to C 1s, O 1s, Zr 3d, Cu 2p, and Cl 2p are clearly observed, confirming the presence of these elements in the composite. Furthermore, XPS spectrum of  $\text{CuCl}_2@G808$  after  $\text{NH}_3$  adsorption is also recorded (Fig. 4). Following  $\text{NH}_3$

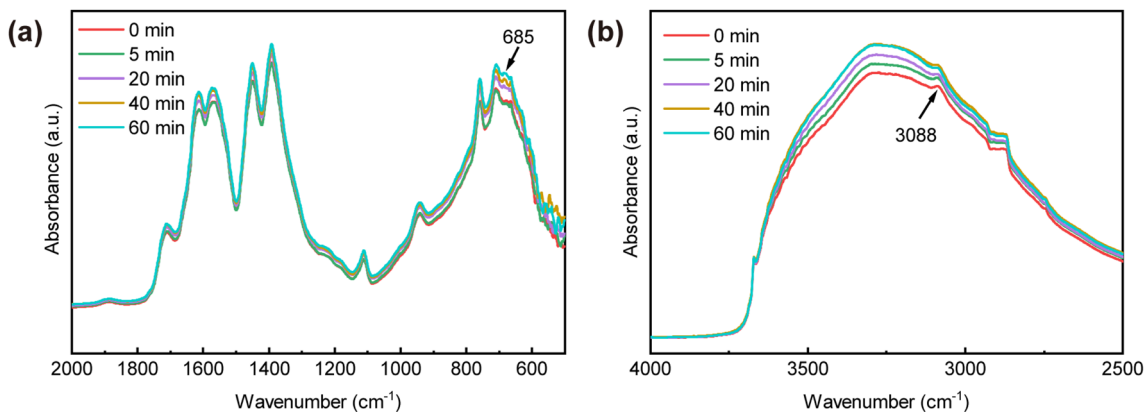


Fig. 3 (a) and (b) *In situ* FTIR spectra of  $\text{CuCl}_2\text{@G808}$  during  $\text{NH}_3$  uptake.

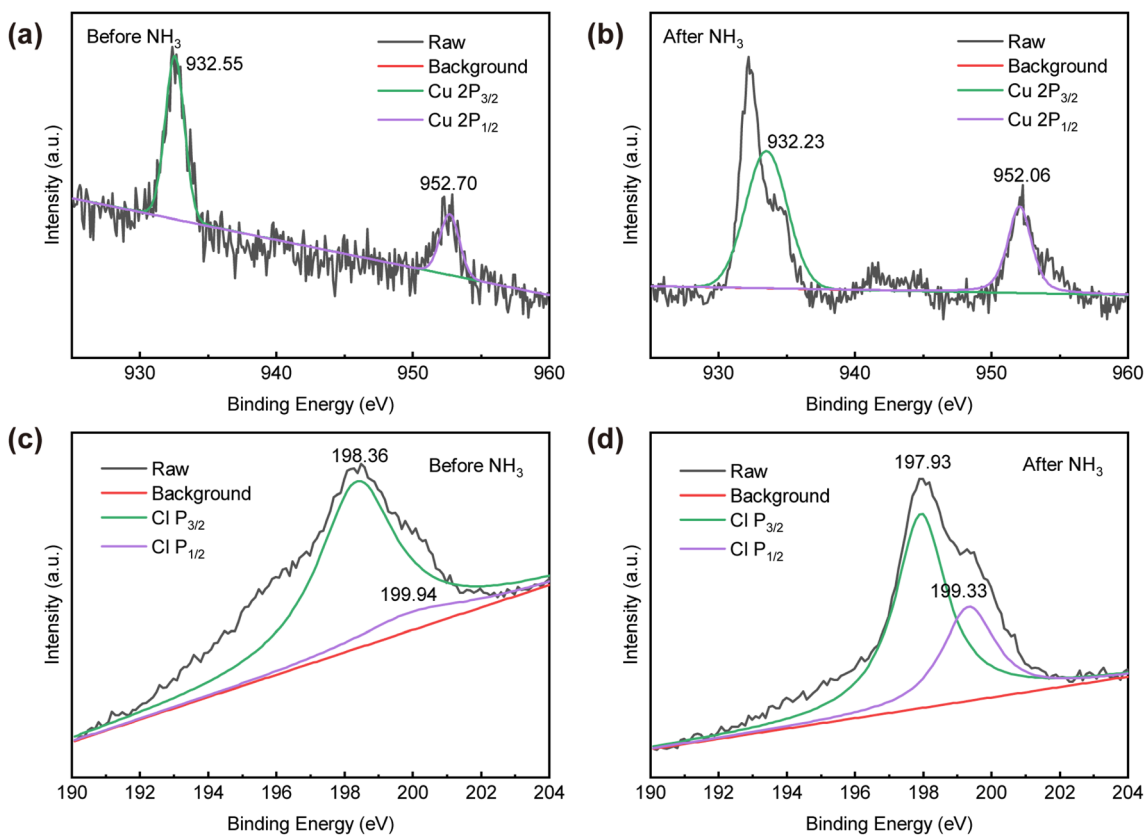


Fig. 4 XPS spectra of Cu 2p (a) and (b) and Cl 2p (c) and (d) for  $\text{CuCl}_2\text{@G808}$  before and after  $\text{NH}_3$  adsorption.

adsorption, the binding energies of Cu  $2p_{3/2}$  and Cu  $2p_{1/2}$  shift from 932.55 eV and 952.70 eV to 932.23 eV and 952.06 eV, respectively (Fig. 4a and b). This negative shift suggests enhanced electron density around  $\text{Cu}^{2+}$  due to coordination with the N atom of  $\text{NH}_3$ , consistent with electron donation from the adsorbate to the metal center.<sup>43</sup> Meanwhile, the Cl 2p peaks of  $\text{CuCl}_2\text{@G808}$  at 199.94 eV and 198.36 eV decrease to 199.33 eV and 197.93 eV, respectively (Fig. 4c and d), attributed to hydrogen bonding between  $\text{NH}_3$  and  $\text{Cl}^-$  sites.<sup>4,43</sup> These findings further validate that  $\text{CuCl}_2$  incorporation enhances the  $\text{NH}_3$  adsorption capacity of the composite at low pressure.

## Conclusion

In summary, we develop a granular  $\text{CuCl}_2\text{@G808}$  metal-organic gel *via* a simple aqueous-phase impregnation method. It is interesting to observe that  $\text{CuCl}_2\text{@G808}$  demonstrates an exceptional  $\text{NH}_3$  uptake ( $2.23 \text{ mmol g}^{-1}$ ) at 298 K and 0.002 bar, which exhibits 79% enhancement over pristine G808 and maintains 67.1% adsorption capacity after five adsorption-desorption cycles. On the other hand,  $\text{CuCl}_2\text{@G808}$  achieves outstanding IAST selectivity for  $\text{NH}_3/\text{N}_2$  ( $2.8 \times 10^3$ ) and  $\text{NH}_3/\text{H}_2$  ( $4.9 \times 10^5$ ). *In situ* FTIR and XPS spectra reveal that low  $\text{NH}_3$



pressure can be mainly explained by  $\text{NH}_3\text{-Cu}^{2+}$  coordination and hydrogen-bond networks. These findings advance the rational design of robust adsorbents for capturing and separating trace  $\text{NH}_3$ .

## Conflicts of interest

The authors declare no conflict of interest.

## Data availability

The authors declare that the data supporting the findings of this study are available within the paper and its supplementary information (SI) files. Should any raw data files be needed in another format they are available from the corresponding author upon reasonable request. Supplementary information is available. See DOI: <https://doi.org/10.1039/d5ra07740k>.

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