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# Organic template-assisted green synthesis of CoMoO<sub>4</sub> nanomaterials for the investigation of energy storage properties†

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Transitional metal oxide nanomaterials are considered to be potential electrode materials for supercapacitors. Therefore, in the past few decades, huge efforts have been devoted towards the sustainable synthesis of metal oxide nanomaterials. Herein, we report a synergistic approach to synthesize spherical-shaped CoMoO<sub>4</sub> electrode materials using an inorganic–organic template *via* the hydrothermal route. As per the synthesis strategy, the precursor solution was reacted with the organic compounds of *E. cognata* to tailor the surface chemistry and morphology of CoMoO<sub>4</sub> by organic species. The modified CoMoO<sub>4</sub> nanomaterials revealed a particle size of 23 nm by X-ray diffraction. Furthermore, the synthesized material was scrutinized by Fourier transform infrared spectroscopy, X-ray photoelectron spectroscopy, field emission scanning electron microscopy and energy dispersive spectroscopy. The optical band gap energy of 3.6 eV was calculated by a Tauc plot. Gas chromatography-mass spectrometry identified cyclobutanol (C<sub>4</sub>H<sub>8</sub>O) and octodrine (C<sub>8</sub>H<sub>19</sub>N) as the major stabilizing agents of the CoMoO<sub>4</sub> nanomaterial. Finally, it was revealed that the bioorganic framework-derived CoMoO<sub>4</sub> electrode exhibited a capacitance of 294 F g<sup>-1</sup> by cyclic voltammetry with a maximum energy density of 7.3 W h kg<sup>-1</sup> and power density of 7227.525 W kg<sup>-1</sup>. Consequently, the nanostructures and organic compounds of *E. cognata* were found to enhance the electrochemical behaviour of the CoMoO<sub>4</sub>-fabricated electrode towards supercapacitor applications.

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## 1. Introduction

One key challenge of the present time is the increasing demand of energy storage devices to store renewable energies.<sup>1–6</sup> Among such storage systems, supercapacitors (SCs) are the candidates with the most potential owing to their high power density in energy storage devices.<sup>5–7</sup> The SCs can efficiently and sustainably fulfill the demands of energy storage as compared to batteries, which are generally of low power density.<sup>7–10</sup> SCs are of different types depending on the nature of the electrode material,<sup>1,6,9,15–18</sup> and their performances are critically reliant on their electrode material.<sup>6,9,15–18</sup> Therefore, in order to achieve high SC performance, an efficient way is to develop electrode materials with tailored and enhanced morphology.<sup>19,20</sup> Over recent years, transition metal oxides have been widely investigated as

electrode materials.<sup>3,6,9–18</sup> The notable advantages of using metal oxides as electrode materials are earth abundance, high theoretical capacity, economical use and ability to be modified in different morphologies and structures.<sup>9–14</sup> In particular, constructing and modifying the morphologies as well as the surface chemistry of the metal oxides at the nanoscale level have been highly investigated among the scientific community.<sup>8–15</sup> The modified nanostructures of transition metal oxides with carbon and oxygen-containing surface functional groups have greatly enhanced the electrochemical performance of SCs.<sup>10–14</sup> Nevertheless, the synthesis of metal oxide nanomaterials with modified surface chemistry has certain limitations such as high cost, large-scale production, sophisticated techniques and complicated synthesis procedures.<sup>1,6,7,9,20</sup> Thus, the sustainable synthesis of metal oxide nanomaterials with effective functional groups is still a great challenge in the 21<sup>st</sup> century.<sup>45,48,51,54,59</sup>

In this regard, herein, we developed a sustainable synthesis route to synthesize metal oxide nanostructures (CoMoO<sub>4</sub>) *via* an inorganic–organic framework. *Euphorbia cognata* Boiss has been used as a source of organic compounds for the first time. The Euphorbiaceae species are reported to have phytochemicals such as diterpenoids, tannins, flavonoids, taraxerol, terpenoids, lectins, cycloartenol, alkaloids, and neriifolins.<sup>74,75</sup> These phytochemicals have been extensively used as reducing and capping agents in the

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synthesis of nanoparticles.<sup>23–28</sup> However, these studies lack an in-depth mechanism and have been unable to identify the incorporated bioactive compounds in the synthesized nanomaterials. Moreover, the reported plant-mediated synthesis is often limited to the mono metal oxides. Therefore, we were motivated to investigate *E. cognata* for the synthesis of CoMoO<sub>4</sub> nanocomposites for the first time. First of all, we have analyzed and identified the organic compounds of the respective plant leaves (shown in S1, S2 and S3†). It revealed different phenolic compounds such as decanoic acid, octodrine, cyclobutanol, D-alanine, and cyclohexylethylamine. Synthetic D-alanine, decanoic acid, and cyclobutanol have been used as fuel and reducing agents in the synthesis of nanomaterials in a number of studies.<sup>21,22,71–73</sup> Therefore, the selection of the plant was made on the basis of its significant phytoconstituents that could be used as reducing and stabilizing agents in the synthesis of nanomaterials. Furthermore, to the best of our knowledge, *E. cognata* has not been previously investigated for the synthesis of nanomaterials.

Plant phytochemicals have been reported as reducing (as well as capping) agents in the synthesis of nanoparticles.<sup>23–28</sup> Thus, we believed that *E. cognata* would induce carbon, oxygen and N-containing organic compounds in the synthesized material. Previously, metal oxides with carbon nanotubes, graphene, and carbon black have been investigated as an outstanding approach for acquiring carbon and oxygen-containing surface functional groups to improve the conductivity and capacitance of the capacitor.<sup>10,11,15,16,19,20</sup> Here, we have tailored the surface chemistry of CoMoO<sub>4</sub> to introduce C, H, O, N-related functional groups *via* organic species of *E. cognata*. We believe that when such functional groups are introduced in the metal oxide nanomaterials, their electronic structure and morphology will be changed, which will create more active sites and quick diffusion pathways. This will lead to better electrical conductivity and an improved redox behaviour of the SC electrode.

## 2. Materials and methods

Molybdenum(II) acetate (Mo<sub>2</sub>(O<sub>2</sub>CCH<sub>3</sub>)<sub>4</sub>), cobalt(II) acetate tetrahydrate (C<sub>4</sub>H<sub>6</sub>CoO<sub>4</sub>·4H<sub>2</sub>O), ethanol (C<sub>2</sub>H<sub>5</sub>OH), methanol

(CH<sub>3</sub>OH), ethylene glycol (C<sub>2</sub>H<sub>6</sub>O<sub>2</sub>) and citric acid (C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>) were purchased from Merck Chemicals, Ltd. Double-distilled and deionized water was used throughout the experiment. The phytochemical extract of *E. cognata* plant leaves was used as reducing and stabilizing agents in the synthesis of CoMoO<sub>4</sub> and was sampled from Rawalakot AJK Pakistan. Acetylene black, polyvinylidenedifluoride (PVDF) and *N*-methyl pyrrolidinone (NMP) were used in the fabrication of the electrode.

### 2.1. Synthesis of CoMoO<sub>4</sub>

The organic–inorganic framework-based strategy to synthesize CoMoO<sub>4</sub> nanomaterials (Fig. 1) was developed by modifying reported methodologies, particularly relating to metal–organic frameworks (MOFs).<sup>19,20,23–28,59,60</sup> 250 ml of 20 mM aqueous solutions of C<sub>4</sub>H<sub>6</sub>CoO<sub>4</sub>·4H<sub>2</sub>O and Mo<sub>2</sub>(O<sub>2</sub>CCH<sub>3</sub>)<sub>4</sub> were prepared separately in deionized water. Each solution was subjected to constant magnetic stirring at room temperature until complete dissolution was achieved; afterwards, both solutions were mixed together. In another beaker, 2 g of dried powdered leaves of *E. cognata* were treated with deionized water on vigorous stirring for 30 minutes at 60 °C to extract organic compounds. After 30 minutes, the extract was cooled down at room temperature and then filtered through Whatman filter paper. Hereinto, 20 ml of the plant organic complex was added into a mixed solution of the precursor on continuous stirring at 70 °C for 2 h, and then incubated at room temperature under dark conditions for 24 hours to attain the phyto-functional groups, and to complete the hydrolysis and precipitation process. Hereafter, the mixture of inorganic metal precursors and organic species of *E. cognata* was first evaporated at 95 °C overnight. The dried powder was then annealed at 450 °C for 4 hours to procure the CoMoO<sub>4</sub> nanomaterial.

The thermally annealed CoMoO<sub>4</sub> nanomaterial was mixed with acetylene black and polyvinylidenedifluoride (PVDF) in an 8 : 1 : 1 ratio in the presence of *N*-methyl pyrrolidinone (NMP) following reported methodologies.<sup>29–32</sup> The prepared slurry was then consistently pasted onto the



Fig. 1 Schematic diagram of the modified hydrothermal method *via* organic compounds to synthesize CoMoO<sub>4</sub> for fabrication of the electrode.



porous Ni foam and dried at 60 °C under vacuum for 10 h. The mass loading was accurately measured by weighing the nickel foam before and after electrode preparation using an analytical balance.

### 3. Characterization

The functionalized  $\text{CoMoO}_4$  was well characterized by ultraviolet-visible spectroscopy (UV-Vis, spectrophotometer 1602, Biomedical services, Spain) for band gap and optical properties. Phyto-stabilizing agents were analyzed by gas chromatography-mass spectroscopy (GC-MS-QP5050, SHIMADZU) and Fourier transforms infrared spectroscopy (FTIR, 8400, Shimadzu, Japan). The phase identification and crystallinity were examined by an XRD5 PANalytical X'Pert Pro. A Quanta 250-FEG scanning electron microscope (FE-SEM) with a Gatan 3View attachment was used to study the morphology and energy-dispersive X-ray spectroscopy (EDX) was used to study the chemical composition of the synthesized  $\text{CoMoO}_4$ . The surface chemistry was investigated by X-ray photoelectron spectroscopy (XPS) using a Kratos Axis Ultra spectrometer with a monochromated Al  $K\alpha$  X-ray source.

### 3.1. Electrochemical measurements

Electrochemical characterizations of the organic compound stabilized  $\text{CoMoO}_4$  were carried out using the three-electrode system, where fabricated nickel foam with  $\text{CoMoO}_4$  was used as the working electrode, a platinum wire was used as the counter electrode and a saturated calomel electrode was used as the reference electrode. All experiments were conducted in 3 M KOH aqueous solution. The electrochemical properties of  $\text{CoMoO}_4$  were investigated by cyclic voltammetry (CV) at various scan rates (2 to 300  $\text{mV s}^{-1}$ ), galvanostatic charge-discharge (GCD) at various current densities (0.5–30  $\text{A g}^{-1}$ ) and electrochemical impedance spectroscopy (EIS) with a frequency range of 50 mHz to 10 kHz.

## 4. Results and discussion

### 4.1. Phyto-functionalized $\text{CoMoO}_4$ nanomaterial

In order to identify the bioactive organic compounds of the plant extract, the spectroscopic characterization of a prepared extract of *E. cognata* leaves was carried out and revealed the distinctive peaks as shown in S1,<sup>†</sup> Fig. 1b. The identified peaks in the ultraviolet and visible regions correspond to plant



Fig. 2 Identification of organic stabilizing agents of  $\text{CoMoO}_4$ . (a) FTIR spectra of  $\text{CoMoO}_4$  after drying at 95 °C. (b) FTIR of  $\text{CoMoO}_4$  after annealing at 450 °C, and (c) GC-MS spectra of  $\text{CoMoO}_4$ .



Table 1 Functional groups of organic compounds of CoMoO<sub>4</sub> identified by recorded peak frequencies via FTIR<sup>a</sup>

Peaks (cm <sup>-1</sup> ) after oven drying	Peaks (cm <sup>-1</sup> ) of annealed sample	Bond	Assignments	Functional groups
3489.69	3499.39	O-H	<i>v</i>	Plant phenolic compounds <sup>23,24</sup>
	2358.46	C=N	<i>v</i>	Nitrile compounds of <i>E. cognata</i> <sup>23,26</sup>
1576.17		C-C (in ring)	<i>v</i>	Aromatic ring of phenol and flavonoids of <i>E. cognata</i> <sup>24-26</sup>
1411.04		C-C (ring)	<i>v</i>	Phenolic compounds of the plant <sup>23-26</sup>
1254.13	1263.89	C-O	<i>v</i>	Phenol and carboxylic acids of the plant leaves <sup>26</sup>
1024.01		C-N	<i>v</i>	Plant amines groups of phytochemicals such as C <sub>8</sub> H <sub>19</sub> N <sup>25,26</sup>
803.59	869.5	=C-H	<i>δ</i>	Hydrocarbon (unsaturated) <sup>24-26</sup>
	766.98	C-H	<i>ρ</i>	Alkane (saturated hydrocarbon) from <i>E. cognata</i> phytochemicals <sup>23-25</sup>
	757.29	C-H	<i>ρ</i>	Alkane (paraffin) saturated hydrocarbon of the leaves extract <sup>24-26</sup>
674.33		N-H	<i>ω</i>	1°, 2° amines of plant phenols <sup>24</sup>
610.07		-C≡C-H:C-H	<i>δ</i>	Unsaturated hydrocarbon such as alkynes <sup>23-26</sup>
	529.14	C-M, M-O	<i>v</i>	Metal oxides (Co-O, Mo-O) <sup>23-26</sup>

<sup>a</sup> *v* = stretching, *δ* = bending, *ρ* = rocking, *ω* = wagging.

phenols, as well as flavonoids comprising a benzene ring with the conjugation of ring A and its configurations. The absorbance values at 297.31 nm and 337.81 nm suggest the presence of plant flavonoids having conjugation of rings B and C. The additional absorption bands at 364.10 and 385.96 nm are indicative of flavones and flavonols in the plant extract.<sup>33</sup>

The dried powdered leaves of *E. cognata* were subjected to scan at the full frequency range (400 to 4000 cm<sup>-1</sup>) of FTIR (Fig. 1a, S1†). FTIR illustrated the vibration modes associated with phytochemicals at frequencies (cm<sup>-1</sup>): 3417.98 (O-H stretching and H bond), 2962.76 (C-H), 2918.40 (C-H), 1639.55 (N-H and -C=C-), 1417.73 (C-C medium stretch (in ring)), 1261.49 (C-H wag (-CH<sub>2</sub>X) bond, C-N and C-O), 1097.53 (C-N stretch), 1024.24 (C-N stretch), 864.14 (=C-H, N-H wag and C-H oop), 800.49 (=C-H, N-H wag and C-H oop) and 536.2 (C-Br). Therefore, in agreement with the UV analysis of the leaves, the FTIR studies illustrated the alcohol, phenol, aromatic, alkanes, 1° and 2° amines, aliphatic amines, carboxylic acid, esters and ether groups as the organic constituents of the investigated plant leaves.

In order to confirm the possible role of the organic compounds of *E. cognata* in the synthesis of the nanomaterials, a CoMoO<sub>4</sub> pellet was first dried at 95 °C and revealed the presence of associated phytochemicals, as shown in Fig. 2a. Afterwards, CoMoO<sub>4</sub> was annealed at 450 °C and subjected to FTIR to identify the functional groups of the incorporated bioactive compounds, as depicted in Fig. 2b. Fig. 2b revealed the presence of a Mo<sup>6</sup> and Co<sup>34,35</sup> metal bond in the fingerprint region (500 to 400 cm<sup>-1</sup>), along with major organic compounds of the *E. cognata* leaves as part of the reducing and stabilizing agents of CoMoO<sub>4</sub>. Table 1 further explains the organic groups of CoMoO<sub>4</sub> corresponding to alcohol and phenol, aldehydes, ester, saturated aliphatics, nitro-compounds (N-O), aliphatic amines, carboxylic acid (O-H bend) and 1°, 2° amines, in agreement with the reported literature.<sup>23-26</sup> It is worth noting that after calcinations, vibration peaks at 1576.17 cm<sup>-1</sup>, 1411.04 cm<sup>-1</sup>, 674.33 cm<sup>-1</sup> and 610.07 cm<sup>-1</sup> seemed to disappear (Table 1, Fig. 2b). This is possibly because of the fact that at higher temperatures (450 °C), some organic compounds were

degraded/transformed into their metabolites. However, after the annealing process, alcohol, phenol, aldehydes, saturated aliphatics and carboxylic acid were found to be present in the synthesized nanomaterial as stabilizing agents, while a vibrational frequency appeared at 521.12 cm<sup>-1</sup> corresponding to the M-O and M-C bonds (M = Co and Mo).

The organic stabilizing agents were further characterized by GC-MS. The results pertaining to the GC-MS analysis of a methanolic extract of *E. cognata* identified numerous compounds at their particular retention times (as shown in the annotated mass spectra on S1,† Fig. 3, Table 1, S2 and S3 in the ESI†). The GC-MS profile of *E. cognata* leaves revealed the presence of octodrine, alanine, cyclobutanol, decanoic acid, azabicyclo, 2-aminodecane and cyclohexylethylamine, as shown in Fig. 1. These bioactive compounds as stabilizing agents were identified in the methanolic suspension of the CoMoO<sub>4</sub> nanocomposite as shown in Fig. 2c. The identified peaks in Fig. 2c correspond to cyclobutanol (C<sub>4</sub>H<sub>8</sub>O) and octodrine (C<sub>8</sub>H<sub>19</sub>N), according to the GC-MS profiling of the NIST library. Therefore, GCMS endorsed the functionalization of CoMoO<sub>4</sub> via bio-organic compounds of *E. cognata*.



Fig. 3 XRD patterns of the CoMoO<sub>4</sub> nanomaterial.







Fig. 4 Elemental analysis of the synthesized nanomaterial using energy dispersive X-ray spectroscopy.

Fig. 3 shows the XRD patterns of the bioorganic compound-derived  $\text{CoMoO}_4$  nanomaterial. The diffraction patterns in Fig. 3 demonstrate well-defined prominent peaks at 2 theta ( $\theta$ ) = 13.2, 19.07, 23.30, 25.45, 26.42, 27.21, 28.32, 29.51, 32.02, 33.59, 36.66, 38.75, 40.13, 41.6, 43.54, 43.99, 45.07, 46.2, 47.05, 48.2, 52.09, 53.47, 54.04, 58.41, 60.42 and 64.35, corresponding to the (001), (−201), (021), (201), (002), (−112), (−311), (310), (−131), (022), (−222), (400), (040), (003), (222), (−422), (−223), (113), (−403), (241), (−133), (−204), (−440), (024), (−424) and (243)  $hkl$  planes, respectively. The peak patterns were exactly matched with  $\text{CoMoO}_4$  of ICSD 00-021-0868. However, a minor shift (less than 0.1%) of the diffraction peaks towards lower diffraction angles was observed for some peaks. This indicates a compression of the crystal lattice due to the stress of organic stabilizing agents. The observed XRD patterns revealed the growth of monoclinic shape  $\text{CoMoO}_4$  with a space group of  $C2/m$  and cell parameters (Å) of  $a$ : 10.21,  $b$ : 9.268,  $c$ : 7.022. The crystallite size of  $\text{CoMoO}_4$  was calculated from the full width half maximum (FWHM) of the peaks using the following Debye-Scherrer's equation:

$$D = (0.9 \times \lambda) / (\beta \cos \theta)$$

where  $\lambda$  = X-ray wavelength and  $\beta$  = full width at half maximum intensity of the peak (in Rad). The average particle size of cobalt molybdate was 23.5 nm. It is noted that no individual peak corresponding to Co,  $\text{Co}_3\text{O}_4$ ,  $\text{MoO}_3$ , or  $\text{Mo}_3\text{O}_4$  was found in Fig. 3. Therefore, the final product was  $\text{CoMoO}_4$  as observed from the peaks pattern of XRD, which was confirmed by EDX (Fig. 4).

The EDX spectrum (Fig. 4) describes the chemical composition of the fabricated material to consist of Co, Mo, O and C. According to the atomic ratios of Co, Mo and O, the nanomaterial is classified as  $\text{CoMoO}_4$ . In agreement with the XRD phase analysis, EDX identified the synthesis of cobalt molybdate with a chemical formula of  $\text{CoMoO}_4$ . Nevertheless, the EDX study demonstrated the presence of a significant atomic percentage of carbon because of the organic functional groups of the stabilizing agents. Continuing with the investigation of the chemical composition,  $\text{CoMoO}_4$  was subjected to XPS analysis to examine the local bonding environment of the synthesized material and to speculate the functional groups related to the bioactive compounds of *E. cognata*. The survey scan in Fig. 5a–d shows the composition of surface to be consistent with C, O, Co and Mo. The presence of Mo, Co,  $\text{O}_2$  and C as major elements of the surface chemistry of the sample is in good agreement with the EDX analysis.

Continuing with the investigation of chemical composition,  $\text{CoMoO}_4$  was subjected to XPS analysis to examine the surface composition of the synthesized material and to verify the presence of organic content of functional groups related to bioactive compounds of *E. cognata* at the surface. All spectra are calibrated on the binding energy scale relative to the C 1s hydrocarbon peak. The narrow XPS scans in Fig. 5(a–d) show the surface to contain C, O, Co and Mo in agreement with the EDX analysis.

The main Co  $2p_{3/2}$  peak is rather broad and centred at a binding energy of 780.7, with a broad satellite feature at a binding energy (B.E.) of  $\sim 787$  eV. The Co  $2p_{3/2}$  peak is fitted with multiplet structure for  $\text{Co}^{2+}$  ions as detailed by Gupta and Sen<sup>37</sup> and Biesinger *et al.*<sup>38</sup> A reasonable agreement is obtained, suggesting the surface Co is mainly in the 2+ state. The





Fig. 5 (a) XPS spectra of Co 2p recorded from CoMoO<sub>4</sub>, (b) spectra of Mo 3d, (c) XPS spectrum of O 1s and (d) spectrum of C 1s of CoMoO<sub>4</sub>.

peak at 796.6 eV is the Co 2p<sub>1/2</sub> peak which arises due to spin-orbit splitting.<sup>38</sup> Fig. 5b shows the Mo 3d<sub>5/2</sub> and Mo 3d<sub>3/2</sub> peaks centred at 232.3 and 235.4 eV, respectively.<sup>61</sup> These peaks must be fitted with two doublets at binding energies of 232.3 eV (3d<sub>5/2</sub>) and 235.4 eV (3d<sub>3/2</sub>) from Mo<sup>6+</sup> in CoMoO<sub>4</sub><sup>43</sup> and 229.7 eV (3d<sub>5/2</sub>) and 233.1 eV (3d<sub>3/2</sub>) from Mo<sup>4+</sup>.<sup>62</sup> As for carbon and oxygen, for samples that have been exposed to air the information is often difficult to interpret. The C 1s peak at 284.8 eV is associated with C–C and C–H species, which may imply that there are compounds arising from the plant extract on the surface. Unfortunately due to some charging of the sample the region at higher binding energy where C–O species would lie was not recorded. The O 1s shows a main peak at an energy of ~530 eV, consistent with metal oxides.<sup>42,44,63,64</sup> However, as in the case of C, since the sample has been exposed to air there will also be contributions to the O 1s from water and hydroxide species.

FE-SEM images of the bioorganic-assisted CoMoO<sub>4</sub> are presented in Fig. 6a–c. The CoMoO<sub>4</sub> nanomaterial was examined at

different magnifications with a well-defined nanostructure. It can be observed in Fig. 6 that at lower magnifications (10 and 5 μm), the spherical particles arranged themselves in quite uniform and regular structures. Thus, FE-SEM depicted the uniformly distributed spherical-shaped bioorganic-derived CoMoO<sub>4</sub> nanomaterials. In order to understand the effects of the bioactive compounds on the morphology of CoMoO<sub>4</sub>, a blank experiment (also called the sol-gel experiment) was carried out involving citric acid (C<sub>6</sub>H<sub>8</sub>O<sub>7</sub>) and ethylene glycol (C<sub>2</sub>H<sub>6</sub>O<sub>2</sub>) in the ethanol (C<sub>2</sub>H<sub>5</sub>OH) solvent. All reducing and gelling agents in the control experiments were of synthetic analytic grade. The details of the controlled experiments are given in the ESI.† It can be observed that the chemical synthesis yielded microstructures (Figure 6d–f), which were larger in size than the organic template-assisted nanomaterial. The ethylene glycol-assisted particles had a nearly cuboidal shape, while the bio-templated particles were spherical in shape. Conversely, the chemically synthesized particles revealed less agglomeration, but with microstructures. The organic framework derived shows





Fig. 6 (a–c) FE-SEM images of the organic template-assisted  $\text{CoMoO}_4$  at different magnifications. (d–f) FE-SEM images of the controlled sample (ethylene glycol assisted  $\text{CoMoO}_4$ ) at different magnifications.

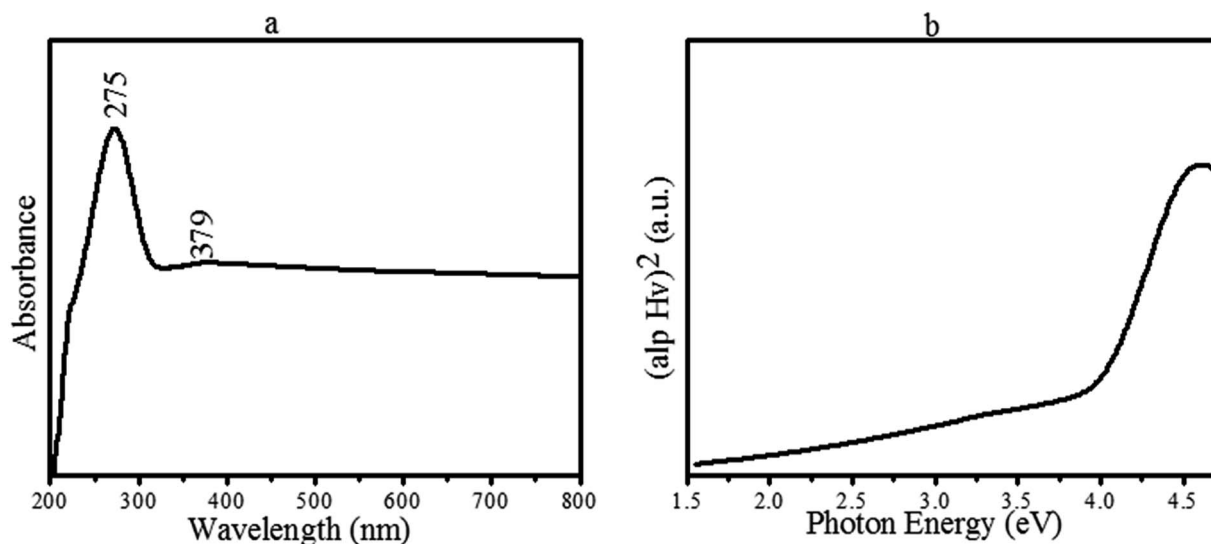


Fig. 7 (a) UV-Vis absorption spectra of  $\text{CoMoO}_4$ , and (b) band gap energy of  $\text{CoMoO}_4$  via Tauc's plot.

little agglomeration at the nanoscale level. Therefore, the biotemplate successfully synthesized nanoparticles, while the chemical synthesis yielded prominent microstructures. Moreover, from the present investigation and reported literature,<sup>23–25</sup> it was concluded that the phytochemicals of the leaves not only reduced the particle size but also prevented the agglomeration of the particles at the nanoscale.

#### 4.2. Band gap energy

Fig. 7a shows the absorption spectrum of the fabricated  $\text{CoMoO}_4$  metal oxides. From Fig. 6a, a wide-range absorption can be observed, which gradually increases towards the lower wavelength with a maximum absorption peak at 275 nm. The maximum absorption band indicates the presence of a blueshift due to the presence of Co oxide, which has a smaller band gap that absorbs the light in a blue shift. However, a small





absorption at 379 nm is also indicated in Fig. 7a. Fig. 7b represents the Tauc plot used to determine the optical band gap energy ( $E_g$ ) of the synthesized material using the following relation:

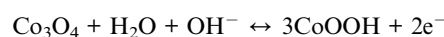
$$(\alpha hv)^2 = A(hv - E_g)$$

where  $\alpha$  is the absorption coefficient, and  $hv$  and  $E_g$  are defined as the photon and band gap energies, respectively. As seen in Fig. 7a, the band absorption edge of CoMoO<sub>4</sub> was found at 349 nm, corresponding to its band-gap excitation. However, the corresponding band gap energy was 3.6 eV for the synthesized CoMoO<sub>4</sub> composite from the Tauc plot.

## 5. Supercapacitive studies

The electrochemical properties of the organic compound-derived CoMoO<sub>4</sub> were characterized by CV, GCD and EIS. Fig. 8a shows typical voltammograms of the fabricated electrode at scan rates ranging from 2 to 300 mVs<sup>-1</sup>. It is worth describing here that in agreement with the previous studies,<sup>24,29,30</sup> the Ni

foam was just used as a mechanical support and current collector in the current investigation. It has no contribution to the capacitance of the electrode because of the negligible integrated area of the CV curves. However, the synthesized CoMoO<sub>4</sub> electrode revealed characteristic redox peaks, as shown in Fig. 8a. Li *et al.* reported that in the CoMoO<sub>4</sub> electrode, the Mo species are not involved in the redox reactions, as its role is to enhance the electrical conductivity of the electrode material. Therefore, according to the reported literature,<sup>9–12,58</sup> the emergence of the prominent redox peaks of the CoMoO<sub>4</sub> electrode are due to the faradaic reactions of M–O and/or M–OH (M = Co), as stated below:



Moreover, the redox peaks also emerged due to the N-containing groups of the incorporated organic compounds



Fig. 8 (a) Cyclic voltammograms of the CoMoO<sub>4</sub> electrode at various scan rates. (b) Galvanostatic charge–discharge characteristics of CoMoO<sub>4</sub> at various current densities. (c) Variation of the specific capacitance as a function of the scan rate, and (d) variation of the specific capacitance as a function of the applied current density.





Table 2 Comparison of the supercapacitive behaviour of the fabricated nanomaterial with reported investigations by cyclic voltammetry

Electrode	Electrolyte	Specific capacitance (F g <sup>-1</sup> )	Scan rate (mV s <sup>-1</sup> )	References
CoMoO <sub>4</sub> @C300	KNO <sub>3</sub>	108.1	10	58
ZnCo <sub>2</sub> O <sub>4</sub> nanowire	PVA/H <sub>3</sub> PO <sub>4</sub>	0.4 mF	100	46
Co <sub>3</sub> O <sub>4</sub> /CB	KOH	52	5	47
MoO <sub>3</sub> -coated TiO <sub>2</sub> nanotubes with four MoO <sub>3</sub> deposition cycles	KCl	51.3	5 mV s <sup>-1</sup>	68
MoO <sub>3</sub> -coated TiO <sub>2</sub> nanotubes with 2 MoO <sub>3</sub> deposition cycles	KCl	44.9	5 mV s <sup>-1</sup>	68
Co-doped TiO <sub>2</sub> NT/RGO	Na <sub>2</sub> SO <sub>4</sub>	34.8	5	49
Pure MoO <sub>3</sub>	KCl	<10	5 mV s <sup>-1</sup>	66
CoMoO <sub>4</sub>	3 M KOH	294	2	Present work
		123	100	
		80.2	300	

(C<sub>8</sub>H<sub>19</sub>N). The organic compounds are reported to have a huge reversible redox potential that makes them interesting candidates for pseudocapacitive energy storage,<sup>36</sup> particularly due to the introduction of nitrogen atoms. Pettong *et al.*,<sup>17</sup> Xie *et al.*<sup>15</sup> and Zhou *et al.*<sup>78</sup> reported in their studies that the introduction of nitrogen atoms greatly improved the pseudocapacitance behaviour of electrodes. In the current study, the redox peaks of CoMoO<sub>4</sub> were found to exhibit a direct positive correlation with the current density and scan rate. The oxidation reduction peaks were found to be increased with increasing current density and increasing the scan rate from 2 to 300 mV s<sup>-1</sup>, as observed in Fig. 8a. The potentials of the redox peaks were shifted accordingly in more positive and negative directions with increasing scan rate, creating a wider potential difference ( $E_c - E_a$ ) at higher potential, under quasi-reversible kinetics.<sup>32,34,49,53</sup> This is possibly attributed to the limited ion diffusion rate during the faradaic reactions to satisfy electronic neutralization at higher scan rates.<sup>13,14,32</sup> This behaviour of the electrode is associated with a typical battery-type electrode.<sup>13,14,32,34</sup> It is further noted that the separation between the redox peaks increased with increasing scan rate, indicating the fast faradaic reactions because of the spherical nanostructure of CoMoO<sub>4</sub>. According to Zhang *et al.* and Bhagwan *et al.*, such redox behaviour is due to the ohmic resistance of the electrode.<sup>4,41</sup> Therefore, the cyclic voltammetry results indicated the functionalized CoMoO<sub>4</sub> as a pseudocapacitor that frequently relied on its redox behaviour as a battery-type electrode. This is because of the presence of Mo, Co, O<sub>2</sub>, and C at the surface of the electrode as major constituents of the active material as shown by XPS in Fig. 3 and by EDX in Fig. 4. The battery-type behaviour is often associated with most metal oxides as reported by Duraisamy *et al.*, Wang *et al.*, and Hussain *et al.*, where they described such pseudocapacitive behaviours of metal oxide-based electrodes.<sup>8,34,53</sup> However, the fabricated CoMoO<sub>4</sub> electrode revealed an enhanced peak current density, as well as a larger increment of the integrated area than MoO<sub>3</sub>,<sup>66</sup> MoO<sub>2</sub>,<sup>65</sup> Co,<sup>8</sup> Co<sub>3</sub>O<sub>4</sub>,<sup>18</sup> and Co<sub>3</sub>O<sub>4</sub>/CoO<sub>34</sub> due to enhanced active sites provided by nanostructures, as well as O and C functional groups. The larger integrated area of the CV curve is

significantly important as it reflects the capacitance of the electrode by the following relation:

$$C_{sp} = \frac{Q}{\Delta V \times \left(\frac{\partial v}{\partial t}\right) \times m}$$

where  $Q$  is the area under the CV curve,  $\partial v/\partial t$  is the scan rate,  $\Delta V$  is the potential window and  $m$  is the mass of the CoMoO<sub>4</sub> on the Ni electrode. As a result, greater the area under redox peaks, higher will be the capacitance.<sup>1,29,30</sup>

As shown in Fig. 8c, the fabricated electrode revealed a higher capacitance of 294 F g<sup>-1</sup> at 2 mV s<sup>-1</sup>, while the specific capacitance was observed to decrease with increasing scan rates because of the inaccessibility of the OH<sup>-</sup> ions of the electrolyte to some parts of the electrode at higher scan rates. Despite this limitation of the electrolyte, a higher capacitance of 122 F g<sup>-1</sup> and 80 F g<sup>-1</sup> were still achievable at the maximum scan rates of 100 and 300 mV s<sup>-1</sup>, respectively. This indicated the excellent rate capability (Fig. 8c) of CoMoO<sub>4</sub> because of a large number of active sites provided by Mo, Co and the functional groups of the organic species, as well as by spherical-shaped nanoparticles. The results of the present study is in agreement with the previous study of Altin *et al.*,<sup>77</sup> where they reported the favourable electrochemistry (considerable specific capacitance) of different ruthenium complexes with four different organic compounds. They reported that the specific capacitance of ruthenium was increased due to the higher surface areas of the organic compounds.

However, the specific capacitance calculated for CoMoO<sub>4</sub> in the current study (even at the highest rate of 300 mV s<sup>-1</sup>) is much higher than the capacitance of 36 F g<sup>-1</sup> of the MoO<sub>3</sub> nanorods,<sup>66,67</sup> 12 F g<sup>-1</sup> of Co<sub>3</sub>O<sub>4</sub>,<sup>47</sup> 77 F g<sup>-1</sup> of the commercial Co<sub>3</sub>O<sub>4</sub> electrode,<sup>18</sup> and 60 F g<sup>-1</sup> of the Co electrode<sup>8</sup> at the lowest scan rate of 5 mV s<sup>-1</sup>. Moreover, Table 2 depicts a comparison of the capacitance of the CoMoO<sub>4</sub> electrode with previously reported electrodes. The comparison vividly demonstrates that the bioorganic compound-derived electrode revealed a much higher potential for the supercapacitor with an outstanding rate capability because of the combined effects of the surface



**Table 3** Comparison of supercapacitive behaviour of CoMoO<sub>4</sub> nanomaterial with reported investigations *via* galvanostatic charge–discharge measurements

Electrode	Electrolyte	Specific capacitance (F g <sup>-1</sup> )	Current density (A g <sup>-1</sup> )	References
Co-doped TiO <sub>2</sub> NT/RGO	Na <sub>2</sub> SO <sub>4</sub>	27.5	0.2	49
Mo-doped ZnO nanoflakes	KOH	46.2	10	50
MnMoO <sub>4</sub>	NaOH	9.7	1	65
CoO/CNT	KOH	17.4	0.25	52
CoMoO <sub>4</sub>	NaOH	62.8	1	65
MoO <sub>3</sub> nanowire	Na <sub>2</sub> SO <sub>4</sub>	23	0.5	69
CoMoO <sub>4</sub>		153.2	0.5	Present work
		97	5	
		49	10	

chemistry and nanoscale morphology analyzed by XPS and FE-SEM, respectively.

The specific capacitance was further calculated from the charge–discharge curve (Fig. 8b) using the following equation:

$$C_{\text{sp}} = \frac{I \times \Delta t}{\Delta V \times m}$$

where  $I$  is the discharge current (A),  $\Delta t$  is the discharge time (s),  $\Delta V$  is the potential window (V), and  $m$  is the mass (g) of the CoMoO<sub>4</sub> nanocomposites on the electrode. The capacitance of CoMoO<sub>4</sub> at various current densities was studied as shown in Fig. 8b and d. The GCD of the CoMoO<sub>4</sub> investigations revealed the highest capacitance of 153 F g<sup>-1</sup> at 0.5 A g<sup>-1</sup>, and the lowest capacitance of 31 F g<sup>-1</sup> at the highest current density of 30 A g<sup>-1</sup>. The calculated capacitance is much greater than the capacitance of 20 F g<sup>-1</sup> of MoO<sub>2</sub>,<sup>70</sup> 23 F g<sup>-1</sup> of the MoO<sub>3</sub> nanowires,<sup>69</sup> and 17.9 F g<sup>-1</sup> of Co/Co<sub>3</sub>O<sub>4</sub> at a lower current density of 1 A g<sup>-1</sup>.<sup>56</sup> In agreement with CV, the GCD measurements revealed an enhanced rate capability of CoMoO<sub>4</sub> due to increased active sites offered by organic compounds. The organic molecules as an electrode material revealed more reactive sites, thus revealing the redox stability that can provide a high specific capacity.<sup>80</sup> These results are consistent with those of Peng *et al.*,<sup>80</sup> Miroshnikov *et al.*,<sup>79</sup> and Xu *et al.*,<sup>81</sup> where it was reported that organic compounds (particularly having carbonyl and/or hydroxyl groups) are mostly derived from biomass, and mainly composed of C, H, O, and N elements, which were positively correlated with increasing capacitance, power and energy density. In another study, Fang *et al.* revealed the excellent specific capacitance (423 F g<sup>-1</sup> at 0.5 A g<sup>-1</sup>) of biomass (betel nut) derived doped carbon. Furthermore, Table 3 summarizes a comparison of the capacitance of the bioorganic framework-synthesized CoMoO<sub>4</sub> with previous investigations, and revealed the comparatively improved potential of the fabricated electrode for the supercapacitor. Therefore, in agreement with the literature,<sup>1,29,30,55,69,70</sup> the current investigations revealed the higher electrode capacitance at lower current densities and lower capacitance at higher current rates, which indicates a battery-type electrode. Moreover, the CoMoO<sub>4</sub>

electrode revealed fast charge and discharge times because of the carbon-based organic stabilizing agents (C=O, C–H) in the sample as revealed by XPS. From Fig. 8b, it can be further observed that the area under the charging and discharging curve is larger at lower current densities and *vice versa*. This indicates the fast charging and discharging of CoMoO<sub>4</sub> at 30 A g<sup>-1</sup>, and the slowest charging and discharging of the electrode at 0.5 A g<sup>-1</sup>. Such inverse correlation between the charge–discharge time and the applied current density proposes that the insufficient faradaic reactions occur due to the poor utilization of the active mass at higher current density.<sup>29–32</sup>

Nevertheless, the presented results from the GCD measurements are in good agreement with the CV results; the fabricated electrode depicted pseudo-capacitive behaviour with outstanding redox peaks. To understand the conducting behaviour of the functionalized CoMoO<sub>4</sub>-based electrodes, EIS measurements were performed for the estimation of the internal resistance and charge transfer resistance. The impedance results are shown in Fig. 9a–c. The Nyquist plot shown in Fig. 9b presented a semicircle arc in the low-frequency region ( $R_{\text{ct}}$ ), and the intercept at the real part in the high frequency range stands for the internal resistance ( $R_i$ ). Fig. 9a shows a line in the low frequency region called the Warburg element ( $Z_w$ ).<sup>1,29,30</sup> It was observed in Fig. 9b that the organic-stabilized CoMoO<sub>4</sub> has a smaller intercept of 0.38 Ω at the real axis in the high frequency region. Thus, it has a smaller  $R_i$  that is due to the nanostructures, as well as the hydrophilic (oxygen) functional groups of the organic species on the surface, as revealed by XPS. The internal resistance of the CoMoO<sub>4</sub> electrodes is smaller than numerous fabricated metal oxide-based electrodes.<sup>8–10,16,32,49</sup> Moreover, Miroshnikov *et al.*<sup>79</sup> and Fang *et al.*<sup>76</sup> demonstrated outstanding capacitive behaviour by EIS analysis of organic compounds and biomass-carbon based electrodes, respectively, due to the C, H, N, and O atoms of the electrode material. Therefore, in the current study, the low internal resistance was strongly attributed to the fact that the pathways for electron transport were not only provided by Mo and Co, but also the organic stabilizing agents (C<sub>4</sub>H<sub>8</sub>O and C<sub>8</sub>H<sub>19</sub>N) of CoMoO<sub>4</sub> as indicated by XPS (Fig. 4) and MS spectra (Fig. 2b).

The fabricated electrode has a low charge-transfer resistance ( $R_{\text{ct}}$ ) according to the semicircle arc in the high frequency region. This is due to the faradaic reactions of Co–Mo–O, as stated by Ujjain *et al.*, Xiao *et al.*, and Zequine *et al.*<sup>16,32,49</sup> Thus,  $R_{\text{ct}}$  confirms the enhanced pseudo-capacitance of the electrode. It can be further observed in the plot of Fig. 9a that there is a clear separation between the semicircle and vertical line of  $Z_w$ . The plot sharply increased in Fig. 9a and nearly became vertical (more close to 45°), corresponding to the  $Z$ -real axis indicating the smaller Warburg impedance. Therefore, the lower  $Z_w$  confirms the rapid diffusion of ions into the electrolyte and fast adsorption onto the electrode surface due to the surface chemistry and connected nanoparticles (Fig. 6) of CoMoO<sub>4</sub>.

Fig. 9c presents the total impedance as a function of frequency. The total impedance of the electrode was observed to be decreased in the present investigation, which is consistent with Gervas *et al.* and Bhoyate *et al.*<sup>29,30</sup> Thus, the lower total





Fig. 9 (a) Nyquist plot of CoMoO<sub>4</sub> in the low frequency region, (b) Nyquist plot (semicircle arc) in the high-frequency region, (c) variation of impedance as a function of frequency, and (d) Ragone plot of CoMoO<sub>4</sub> (inset depicts the response of the energy density at higher power density).

impedance further designated the better conductive characteristics of CoMoO<sub>4</sub>. These results indicated the good electrical conductivity of the fabricated electrode with faradaic reactions. Finally, the storage potential of the fabricated electrode was estimated by the energy density ( $E$ ) and power density ( $P$ ) using the discharge voltage *via* the following equations and are expressed as  $W\ h\ kg^{-1}$  and  $W\ kg^{-1}$ :

$$\text{Energy density } (E) = \frac{C \times \Delta V^2}{7.2}$$

$$\text{Power density } (P) = \frac{E \times 3600}{t}$$

where  $C$  ( $F\ g^{-1}$ ) is the capacitance *via* galvanostatic charge discharge,  $\Delta V$  (V) is the potential window and  $t$  (s) is the discharge time.<sup>29,30,56</sup> In the present study, the maximum calculated energy density of CoMoO<sub>4</sub> was 7.3 against a power density of 146, which is remarkably higher than the energy density of MoO<sub>3</sub>,<sup>66,70</sup> Co<sub>3</sub>O<sub>4</sub> nanorods,<sup>57</sup> CoO/Co<sub>3</sub>O<sub>4</sub>,<sup>49</sup> CoO nanofibers<sup>56</sup> and Co<sub>3</sub>O<sub>4</sub>@Carbon,<sup>56</sup> respectively. The phytofabricated CoMoO<sub>4</sub> exhibited an enhanced energy storage potential in comparison with several reported devices.<sup>39,40,49,56,65–70</sup> On the other hand, the

maximum recorded power density of CoMoO<sub>4</sub> was 7272.5  $W\ kg^{-1}$  (7.3  $kW\ kg^{-1}$ ), which is much higher than the power densities of MoO<sub>3</sub>,<sup>66,70</sup> Co<sub>3</sub>O<sub>4</sub> nanorods,<sup>57</sup> CoO/Co<sub>3</sub>O<sub>4</sub>,<sup>49</sup> CoO nanofibers<sup>56</sup> and Co<sub>3</sub>O<sub>4</sub>@Carbon,<sup>56</sup> respectively. The Ragone plot (Fig. 9d) shows that when the power density increased, the energy density decreased and *vice versa*. A negative correlation between the energy density and powder density is a general phenomena associated with numerous other fabricated supercapacitors.<sup>29–32</sup> The inset in the Ragone plot (Fig. 9d inset) revealed the energy response of CoMoO<sub>4</sub> at the higher power densities. It is worth reporting that, in the current research, the energy density remained at 1.85  $W\ h\ kg^{-1}$  at the higher power density of 4  $kW\ kg^{-1}$ . This lower energy density of CoMoO<sub>4</sub> is still higher than the energy densities of Co@Carbon//Co@Carbon and Co<sub>3</sub>O<sub>4</sub>@Carbon//Co<sub>3</sub>O<sub>4</sub>@Carbon.<sup>56</sup> Consequently, in comparison with the previously reported investigation, the energy density and power density strongly suggested the potential of CoMoO<sub>4</sub> as an energy storage device.

Thus, in agreement with the low resistance, outstanding rate capability and efficient redox behaviour, the energy density and power density demonstrated the possibility of the fabricated CoMoO<sub>4</sub> electrode as a potential candidate for a supercapacitor.



## 6. Conclusion

Hereinto, we have successfully developed an inorganic–organic template for the synthesis of nanospherical-shaped cobalt molybdate using bioactive compounds of *E. cognata*. The obtained organic compound-assisted CoMoO<sub>4</sub> not only revealed the stabilization by octodrine and cyclobutanol, but also facilitated the enhanced electro-active sites for energy storage devices. CoMoO<sub>4</sub> demonstrated its efficient redox behaviour with a specific capacitance of almost 300 F g<sup>-1</sup> in 3 M KOH aqueous solution. Furthermore, significant capacitance values of 123 and 61.2 F g<sup>-1</sup> were still retained at a higher scan rate of 100 mV s<sup>-1</sup> and current density of 10 A g<sup>-1</sup>, respectively. Such excellent rate capability has been ascribed to the synergetic effects of the nanostructures and increased active sites created by C and O-related functional groups. In addition, the pathways provided by the hydrophilic functional oxygen groups of the organic stabilizing compounds, have reduced the resistance and enhanced diffusion/transport of electrons. Consequently, the considerable specific capacitance, good rate capability, and enhanced electrochemical conductivity promote the potential application of bioorganic compound-fabricated CoMoO<sub>4</sub> nanomaterials for supercapacitors.

## Conflicts of interest

The authors declare that they have no known conflict of interest that would influence the work reported in this paper.

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