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Supracluster assembly of an Al₄ precursor toward the study of enhanced optical properties†

Fan Yang, ‡^{a,b} Rui-Yan Chen, ‡^{a,c} San-Tai Wang, ^{a,b} Yan-Ping He, ^b *^a Wei-Hui Fang ^b *^{a,b} and Jian Zhang ^a

Supracluster assembly offers a unique opportunity for the isolation of customized materials using functional cluster precursors. Herein, for the first time, we demonstrate supracluster assembly based on an Al₄⁴⁻ precursor through both non-covalent interactions and coordination bonds. By reacting readily available, ultrastable, and soluble Al₄⁴⁻ as an anionic precursor with a metal center and neutral chelated chromophores, we successfully synthesized an extensive library of supracluster assembled crystalline solids, ranging from ion-pair structures, surface-modified clusters, and chains, to layered compounds. The advantage of this method is that it achieves stepwise performance regulation and enhancement using given functional precursors. These materials from supracluster assembly possess a broad range of tunable photophysical properties, including photoluminescence and third-order nonlinear optical (NLO) properties. Such extraordinary supracluster assembly will be widely applicable for creating a new generation of rationally organized solid materials with tailorable and customized properties.

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Introduction

Post-synthetic modification is a useful technique to precisely prepare new target materials by introducing desired properties into pre-designed structures. Such a method involves introducing small organic molecules or polymers, metal ions, *etc.* into pre-synthesized materials without destroying the structure of the material itself so as to obtain new functional materials. Therefore, it is prevalent and widely used in a broad range of materials, including metal-and covalent organic frameworks, 1,3-5 porous- and metal-organic cages, 2,6 and interlocked molecules.

The molecular nature of discrete metal-organic cages and cluster compounds endows them with easy solution-processability unattainable with traditional framework materials.⁸ Thus, these discrete compounds can be adapted for stepwise assembly through solution processing in addition to the solid state. In this regard, the stepwise approach involves dissolving these discrete supramolecular entities in an appropriate

Recently, we reported the supracluster assembly of Archimedean {Al₆M₃}₈ cages with 72 hydrogen bonds.³⁶ The supracluster assembly fills the gap between "supramolecular"37 and "superparticle"38 systems. The word "supracluster" is rarely used but was once used to describe and emphasize the size of clusters, where Teo and Dale named compounds $[(Ph_3P)_{10}Au_{12}Ag_{12}PtCl_7]Cl$ and $[Ru_6C(CO)_{17}]$ in 1993 and 1995, ^{39,40} whereas our proposed supracluster assembly aims to achieve "chemistry beyond the cluster", in that attention should be paid to the interactions between clusters, such as hydrogen bonds, $\pi \cdots \pi$ interactions and covalent bonds, with the interactions between the vertex and edge-sharing supracluster assemblies based on Archimedean {Al₆M₃}₈ cages being hydrogen bonds. In our previous work, we investigated the coordination-driven self-assembly of aluminum molecular rings, which can be viewed as supracluster assembly via coordination bonds.

Herein, for the first time, we demonstrate supracluster assembly based on an Al₄ precursor through both hydrogen

^aState Key Laboratory of Structural Chemistry, Fujian Institute of Research on the Structure of Matter, Chinese Academy of Sciences, 350002 Fuzhou, P.R. China. E-mail: hyp041@163.com, fwh@fjirsm.ac.cn

^bUniversity of Chinese Academy of Sciences, Beijing 100049, China

^cMOE Key Laboratory of Cluster Science, School of Chemistry and Chemical Engineering, Beijing Institute of Technology, Beijing 100081, China

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[‡]These authors contributed equally to this work.

bonds and coordination bonds. Based on our recent efforts and progress in aluminum oxo clusters (AlOCs), 41-47 we believe that $(Me_2NH_2)_4[Al_4(L)_4(Cat)_2]$ (AlOC-13, a cluster denoted as Al₄ hereinafter)⁴² is a suitable precursor for further study of supracluster assembly of target materials with photorelated properties (Scheme 1) where (1) it fulfills the requirements of high solubility/stability and accessibility for scale-up synthesis; and (2) it provides one successful example of incorporating lanthanides⁴² proving the possibility of assembly and that the abundance of interaction sites (aromatic ring, hydroxyl group and bridging oxo) will bring more interesting modes of assembly. The presence of aromatic rings and an oxo moiety may lead to weak interaction with guest molecules. The suspended oxo group may result in coordination with oxyphilic metal ions. Such assembly is feasible from the point of view of charge matching. We selected a series of chelating ligands of different lengths, and the distance between the two chelating sites plays a key role in assembling the compound type to form ion pairs and coordination-modified compounds, respectively. With subsequent assembly, the photo-related properties will also be tuned. Accordingly, we probe an avenue toward the enhancement of intended optical properties like photoluminescence and third-order nonlinear optical (NLO) effects. For example, the employment of chiral ligands and heavy metal ions enhances third-order NLO performance. 48,49

Results and discussion

Synthesis

The facile synthesis of large amounts of AlOC-13 as the source of the Al₄ clusters is described in detail in the Experimental section (Scheme S1 and Tables S1-S3†). It is a monodisperse anionic cluster with (Me₂NH₂)⁺ countercations, which can be soluble and stable in water and common solvents. There are eight uncoordinated carboxyl oxygen atoms and six phenyl rings in each Al4 cluster, making it an excellent precursor for assembly with cationic units. The reasons we chose the N,N- chelated (phenanthroline-type) and P,P-chelated (phenylphosphine-type) ligands are that they meet the requirements of both ligand chelate and photo-functional conjugated groups (Fig. 1). By investigating the supracluster assembly behaviors of Al₄ clusters with metal ions (such as Ag⁺, Fe²⁺ and Cu²⁺), we successfully obtained a variety of novel ionic pair compounds via short chelating ligands (hydrogen bonds and $\pi \cdots \pi$ interactions) (AlOC-168 to AlOC-172) and coordination-modified compounds through relatively long chelating ligands (AlOC-173 to AlOC-176) (Table 1). Rapid crystal growth of these materials can be achieved by solvent-thermal synthesis in one day.

Supracluster assembly of structures *via* non-covalent bonds

In the beginning, AgClO₄ and the 1,3-bdpp ligand were added to a DMF/EtOH/H2O solution of the Al4 precursor, which was kept at 80 °C for one day, yielding faint yellow block crystals of AlOC-168. In the structure of AlOC-168, the Ag(I) center is P,Pchelated by two 1,3-bdpp ligands, giving rise to an [Ag(1,3bdpp)₂]⁺ ion (Fig. S1†). The supracluster assembly of the

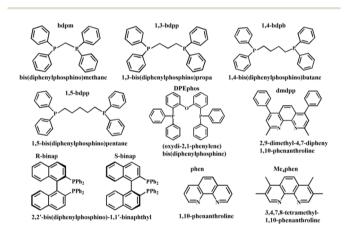
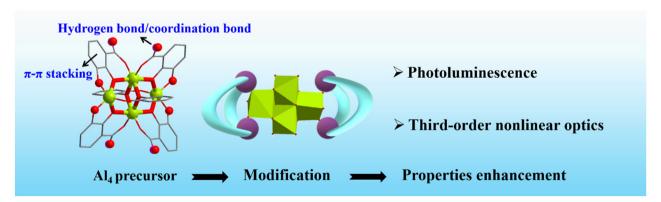


Fig. 1 The N,N-chelated or P,P-chelated ligands used in the supramolecular assembly based on Al₄ clusters.



Scheme 1 Illustration of the selection of the Al₄ precursor and its supracluster assembly towards enhancement of photo-related properties. The advantages of using the Al₄ precursor include its high solubility/stability and accessibility for scale-up synthesis, as well as abundant binding sites. The supracluster assembly method includes non-covalent interaction assembly and coordination bond interaction assembly. The dark purple balls represent metal centers like Ag⁺, Fe²⁺ and Cu²⁺, while the cyan loops represent flexible ligands.

Table 1 Summary of the compounds resulting from the supracluster assembly of the Al₄ precursor

Complex	Composition	Space group	Structure dimensionality	R_1 value	CCDC			
Supracluster assembly of structures <i>via</i> non-covalent bonds								
AlOC-168	$(Me_2NH_2)_3[Ag(1,3-bdpp)_2][Al_4(L)_4(Cat)_2]$	$P\bar{1}$	0D	0.053	2303391			
AlOC-169	$(Me_2NH_2)_3[Ag(dmdpp)(DPEphos)][Al_4(L)_4(Cat)_2]$	Pbca	0D	0.138	2303392			
AlOC-170(R)	$(Me_2NH_2)_6[Ag(R-binap)_2]_2[Al_4(L)_4(Cat)_2]_2$	$P2_1$	0D	0.077	2303393			
AlOC-170(S)	$(Me_2NH_2)_6[Ag(S-binap)_2]_2[Al_4(L)_4(Cat)_2]_2$	$P2_1$	0D	0.053	2303394			
AlOC-171	$[Fe(phen)_3]_2[Al_4(L)_4(Cat)_2]$	I2/a	0D	0.041	2303395			
AlOC-172	$\left[\text{Cu}(\text{Me}_4\text{phen})_2(\text{H}_2\text{O})\right]_2\left[\text{Al}_4(\text{L})_4(\text{Cat})_2\right]$	$P\bar{1}$	0D	0.051	2303396			
Supracluster assembly of structures via coordination bonds								
AlOC-173	$[Ag_2(bdpm)_2]_2[Al_4(L)_4(Cat)_2]$	$P\bar{1}$	0D	0.053	2303397			
AlOC-174	$[Ag_3(1,5-bdpp)_3]_2[Al_4(L)_4(Cat)_2](ClO_4)_2$	$P\bar{1}$	0D	0.043	2303398			
AlOC-175	$[Ag_4(1,4-bdpb)_3][Al_4(L)_4(Cat)_2]$	I2/a	1D	0.036	2303399			
AlOC-176	$[Ag_4(bdpm)_2(1,4-bdpb)][Al_4(L)_4(Cat)_2]$	$Par{1}$	2D	0.042	2303400			

H₃L = 2,3-dihydroxybenzoic acid; cat = catechol; bdpm = bis(diphenylphosphino)methane; 1,3-bdpp = 1,3-bis(diphenylphosphino)propane; 1,4bdpb = 1,4-bis(diphenylphosphino)butane; 1,5-bdpp = 1,5-bis(diphenylphosphino)pentane; DPEphos = (oxydi-2,1-phenylene)bis(diphenylphosphine); R-binap = (R)-(+)-2,2'-bis(diphenylphosphino)-1,1'-binaphthyl; S-binap = (S)-(-)-2,2'-bis(diphenylphosphino)-1,1'-binaphthyl; phen = 1,10-phenanthroline; dmdpp = 2,9-dimethyl-4,7-diphenyl-1,10-phenanthroline; Me4phen = 3,4,7,8-tetramethyl-1,10-phenanthroline.

anionic Al₄ cluster and cationic [Ag(1,3-bdpp)₂]⁺ unit forms an interesting ion-pair structure (Fig. 2a). However, in the above synthetic system of AlOC-168, we replaced the 1,3-bdpp ligand with DPEphos and added the dmdpp phenanthroline-type ligand, and we readily obtained another ion-pair structure (AlOC-169) composed of an Al₄ cluster and [Ag(dmdpp) (DPEphos)]+ unit (Fig. S4†), in which the Ag(I) center is chelated by one DPEphos ligand and one dmdpp ligand. When we replaced the organic ligand with a chiral binap ligand, we successfully synthesized a pair of pure enantiomers (AlOC-170(R)

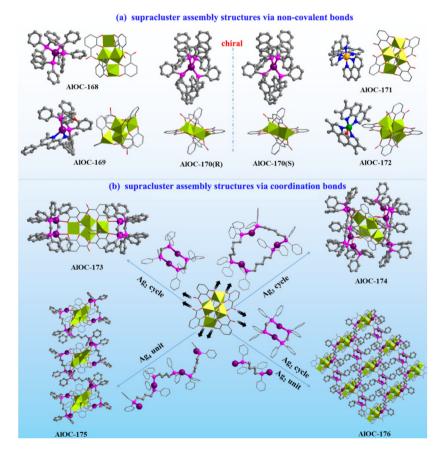


Fig. 2 A family of structures from supracluster assembly based on pre-designed Al₄ clusters including (a) supracluster assembly of structures via non-covalent bonds and (b) supracluster assembly of structures via coordination bonds. Atom color code: green, Al; modena, Ag; orange, Fe; olive, Cu; pink, P; red, O; blue, N; gray, C; off-white, H.

and AlOC-170(S)) (Fig. S7 and S10†). In their structures, two Ror S-binap ligands coordinate to one Ag(1) center, producing a chiral $[Ag(R-binap)_2]^+$ or $[Ag(S-binap)_2]^+$ unit, which further forms chiral ion-pair structures with the Al4 cluster. Their absolute configuration and enantiomeric nature have been confirmed by circular dichroism (CD) spectra, whose CD peaks exhibit obvious mirror images (Fig. S61†). Considering the chiral and fluorescent properties of AlOC-170R/S, 50,51 we tested their circularly polarized luminescence, and the results were not as expected. In the 3D packing structures of AlOC-168 to AlOC-170, there are multiple supramolecular interactions, including $\pi \cdots \pi$ stacking and C-H··· π , C-H···O, O-H···O, and N-H···O interactions (Fig. S3, S6 and S9†). It is particularly necessary to point out that the Ag units form a cationic layer, while the Al₄ clusters form an anionic layer (Fig. S2, S5 and S8†), and they take the ABAB alternating stacking mode, which is similar to that of hydrotalcite. Furthermore, the (Me₂NH₂)⁺ countercations are located in the anionic Al4 cluster layer. By replacing the above AgClO4 with FeSO4·7H2O and Cu2O and adding a small amount of ammonium hydroxide, we synthesized ion-pair structures AlOC-171 and AlOC-172 in the presence of phen and Me₄phen ligands, respectively. The Fe(II) and Cu(II) centers are N,N-chelated by two phen and Me4phen ligands (Fig. S11 and S13†), respectively, and there is also an H₂O molecule that coordinates to the Cu(II) center. Through weak interactions, the Al₄ clusters and the $[Fe(phen)_3]^{2+}$ or $[Cu]_4$ $(Me_4phen)_2(H_2O)$ ²⁺ cations stack alternately into a 3D dense structure (Fig. S12 and S14†).

Research Article

Supracluster assembly of structures via coordination bonds

Encouraged by the above results, more assembly experiments were explored (Fig. 2b). When increasing the dosage of Ag ions and using the shorter bdpm ligand, we synthesized AlOC-173. In the structure of AlOC-173 (Fig. S15 and S16†), two Ag(I) centers are connected by two bdpm ligands, in situ forming a binuclear [Ag₂(bdpm)₂]²⁺ cycle (named the Ag₂ cycle). Interestingly, each Al₄ cluster links two such Ag₂ cycles via the Ag-O coordination bond on both sides of the cluster, in which the carboxyl O atoms of the Al₄ cluster coordinate to the Ag(1) atom of the Ag₂ cycle. The Ag···Ag length is about 2.9 Å, and the Ag-O bond lengths are about 2.5 Å, which are within the normal bond-length range. All the Ag(1) centers are three-coordinated. As we continued to increase the dosage of Ag⁺ ions and used a longer 1,5-bdpp ligand, we obtained AlOC-174. Structural analyses show that each Al₄ cluster in **Aloc-174** traps two in situ formed $[Ag_3(1,5-bdpp)_3]^{3+}$ cycles (each named as an Ag₃ cycle). As shown in Fig. S17,† the Ag1 atom is four-coordinated, while Ag2 and Ag3 atoms are threecoordinated. The (ClO₄) counteranions are present in the structure of AlOC-174 (Fig. S18†). Herein, through coordination interactions, we modified the Ag₂ and Ag₃ cycles onto the Al₄ cluster and obtained two functional ligand-modified heterometallic clusters.

From the above results, it can be seen that the length of the organic ligand also affected the supracluster assembly behavior of the Al₄ cluster, so we continued to change the type of

ligand and tried using mixed organic ligands. When the 1,4bdpb ligand was used to react with Ag⁺ ions and Al₄ clusters, AlOC-175 was synthesized. In AlOC-175, three 1,4-bdpb ligands link with four Ag(1) centers to generate a tetranuclear [Ag₄(1,4bdpb)₃]⁴⁺ unit (named the Ag₄ unit) (Fig. S19†). Such an *in situ* generated Ag₄ unit further connects two adjacent Al₄ clusters through Ag-O coordination bonds, giving rise to an infinite chain structure. The chains are stacked in ABAB mode to form a dense supramolecular structure (Fig. S20†), and there are rich interactions between adjacent chains in AlOC-175, including $\pi \cdots \pi$ stacking, C-H··· π and C-H···O interactions. However, when longer 1,4-bdpb and shorter bdpm ligands were added to the reaction system simultaneously, we obtained compound AlOC-176. Structural analysis shows that AlOC-176 has a layered structure of Al₄ clusters linked together by $[Ag_2(bdpm)_2]^{2+}$ cycles (named the Ag_2 cycle) and $[Ag_2(1,4$ bdpb)]²⁺ units (named the Ag₂ unit) (Fig. S21†). In the structure of AlOC-173, the Ag2 cycle is only modified at both ends of the cluster, while it acts as a linker in AlOC-176, collaborating with the Ag2 unit to further connect with and form the Al4 clusters into a layer (Fig. S22†).

Herein, we demonstrated an extraordinary supracluster assembly including both non-covalent and coordination bonds. These assembled compounds are challenging to synthesize in one pot, so through supracluster assembly, we can indeed obtain new materials that cannot be obtained by direct synthesis. In theory, all kinds of metal cation units can form ion-pair structures with the anionic Al₄ cluster. Similarly, coordination bonds can also modify different metal cation units at the face of the cluster. On the other hand, the Al₄ cluster can also be organized into different dimensional structures, including chain and layer structures, as well as 3D frameworks. Through functional modification, we can obtain the desired applications.

The powder X-ray diffraction (XRD) patterns of AlOC-168 to AlOC-176 are shown in Fig. S23-S31.† To ensure the high purity of all structures, we performed Le Bail fitting. The low $R_{\rm wp}$ and $R_{\rm p}$ values (<10%) in the Le Bail fittings show that these compounds have high phase purity, indicating that the self-assembly of Al₄ clusters with metal cation units into these cluster-based compounds under certain conditions is quite stable and directed. The thermogravimetric analysis (TGA) curves of these materials were recorded, as shown in Fig. S32-S40 in the ESI.† Their TG curves do not have a relatively apparent weightless platform before 300 °C, which is consistent with their dense supramolecular packing structures. The solvents or counterions $((Me_2NH_2)^+ \text{ or } (ClO_4)^-)$ were gradually lost with increasing temperature, followed by decomposition of their structures. The residues after decomposition might be Ag₂O and CuO or Fe₃O₄. Diffuse reflectance spectroscopy was used to study the UV/vis absorption of these compounds. According to the Kubelka-Munk function, yellowish Agcluster- and green Cu-cluster-based compounds exhibit relatively low bandgaps (2.6-3.1 eV) compared to that of the Al₄ cluster raw material (3.2 eV), but that of the deep red **Aloc-171**, containing the $[Fe(phen)_3]^{2+}$ unit, is about 1.56 eV,

which is significantly lower than that of other compounds (Fig. 3a, b and S42-S50†). Their homogeneous elemental distribution was confirmed by their spectra from energy dispersive spectroscopy (EDS) as shown in Fig. S51-S58.†

Studies of photoluminescence properties

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Because the phenanthroline- and phenylphosphine-type ligands possess strong absorbing chromophores, and the Agbased compounds themselves have good photoluminescent properties, we decided to study the solid-state excitation and emission spectra of compounds AlOC-168 to AlOC-176 at room temperature (Fig. 3c and d). For convenience, their detailed parameters are also summarized in Table 2. As shown in Fig. S59,† upon excitation at 400 nm, the Al₄ precursor raw material displays an emission band at 400-700 nm and the highest emission peak at 487 nm, which may belong to the π $\rightarrow \pi^*$ transition of organic ligands. Under the same excitation conditions, these cluster-assembled compounds AlOC-168 to AlOC-170 and AlOC-173 to AlOC-176 show similar emission peaks, which can be observed at 500, 520, 489, 533, 477, 505 and 510 nm, respectively. However, AlOC-171 and AlOC-172 have no obvious photoluminescence under ultraviolet light (360 nm). Compound AlOC-174 exhibits a blue-shift (ca. 10 nm) in comparison with the Al₄ cluster, and it shows blue light. However, AlOC-173 and AlOC-176 display distinct redshifts of ca. 20-50 nm, and their luminescent colours are viridescent and yellow, respectively (Fig. S60†). In these case, the ligand-based luminescence is dominant and the blue-shifted or red-shifted phenomenon may be attributed to metal-ligand/ cluster coordinative interactions. Additionally, the rigidity of

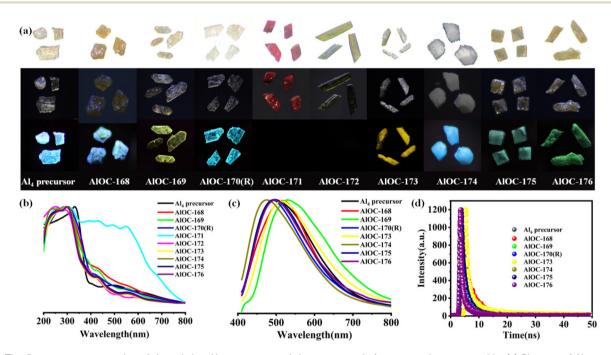


Fig. 3 The fluorescence properties of the pristine Al₄ precursor and the compounds from supracluster assembly. (a) Pictures of Al₄ precursor to AlOC-176 crystals. The three images in each column from top to bottom show crystals in natural light, crystals in the dark without fluorescence irradiation, and crystals in the dark with fluorescence irradiation. (b) The UV/vis wavelengths of the Al₄ precursor to AlOC-176. (c) The emission wavelengths of the Al₄ precursor to AlOC-176 except for those of AlOC-171 and AlOC-172. (d) The transient fluorescence lifetime curves of the Al₄ precursor to AlOC-176 except for those of AlOC-171 and AlOC-172.

Table 2 The crucial photophysical parameters of the Al₄ precursor and compounds from supracluster assembly

Compound	UV/vis wavelength (nm)	Excitation wavelength (nm)	Emission wavelength (nm)	Fluorescence lifetime (ns)	β (cm GW ⁻¹)
Al ₄ precursor	332	395	495	2.45	98
AlOC-168	288	395	515	3.24	200
AlOC-169	292	395	525	4.33	120
Aloc-170(R)	302	395	490	4.01	220
AlOC-171	306	395	455	0	130
AlOC-172	254	395	455	0	700
AlOC-173	301	395	515	4.25	750
AlOC-174	306	395	475	2.07	730
AlOC-175	300	395	495	2.02	710
AlOC-176	299	395	500	2.07	700

the ligands in the structure may also account for the fluorescence differences. The rigidity of the ligands in AlOC-173 and AlOC-169 is greater compared to that in other structures, and hence their fluorescence lifetimes are somewhat superior. In a word, the luminescent colour changes can be realized by different supracluster assembly behaviours.

Studies of third-order nonlinear optical (NLO) properties

Research Article

Third-order nonlinear optical (NLO) materials are of particular importance for various applications including the optical limiting effect. It is reported that effects of larger conjugated systems and heavy metals have been proven to have positive influences on third-order nonlinear optical and optical limiting behavior. 49,52-54 Considering the presence of heavy metals and large conjugated chromophores, we then decided to study their third-order NLO properties. In order to ensure the stability of compounds AlOC-168 to AlOC-176, the crystals of these materials are effectively dispersed into polydimethylsiloxanes (PDMSs) and then made into a thin film (named AloCs@PDMS film), respectively. (The Experimental section provides detailed experimental information.) For better comparison, we also tested the third-order NLO properties of the starting Al₄ precursor by the same film-making method (Al₄ precursor@PDMS). The third-order NLO properties of these films were studied using a typical open-aperture Z-scan system with a nanosecond laser at 532 nm.

Precise control of the intramolecular coordination bonds and intermolecular non-covalent forces of clusters tunes

charge transfer and band gap enhancement and ultimately improves the third-order nonlinear optical performance. Under an input laser pulse energy of 80 μJ, Al₄ precursor@PDMS to AlOCs@PDMS display obvious reversesaturated absorption (RSA) behaviors and exhibit a sequentially enhanced optical limiting response from AlOC-168 to AlOC-176 (Fig. 4a and Table S4†). In order to quantitatively evaluate the NLO performance of the sample, the nonlinear absorption coefficient (β) (eqn (1)) of the sample was obtained from the best-fit curve:

$$T(z,s=1) = \frac{1}{\sqrt{\pi}q_0(z,0)} \int_{-\infty}^{\infty} \ln[1 + q_0(z,0)e^{-r^2}] dr \qquad (1)$$

Here, $q_0(z,0) = \beta I_0 L_{\text{eff}}$, where I_0 is the on-axis peak intensity at the focus (z = 0), $L_{\text{eff}} = [1 - \exp(-\alpha l)]/\alpha$ denotes the effective thickness of the sample, α is the linear absorption coefficient, and l indicates the sample thickness. The values of β for Al₄ precursor@PDMS and AlOCs@PDMS composite films are 98, 200, 120, 220, 130, 700, 750, 730, 710, and 700 cm GW⁻¹, respectively (Fig. 4d). Compared with the Al₄ precursor, the introduced large conjugated ligands facilitate charge transfer between anion and cation groups, thus improving the thirdorder NLO performance. In the non-covalently assembled compounds (AlOC-168 to AlOC-172), AlOC-172 has relatively good nonlinear absorption due to its low bandgap and heavy metal effect. As for supracluster assembly via coordination bonds, compounds AlOC-173 to AlOC-176 exhibit better optical limiting responses, which is attributed to the coordination inter-

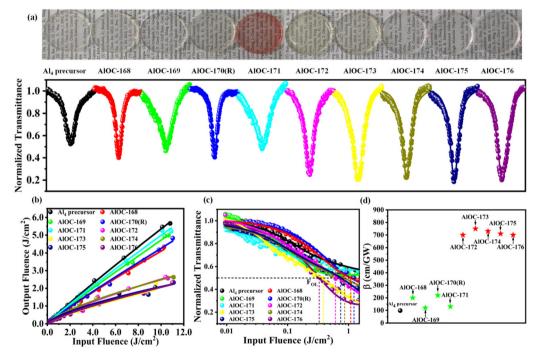


Fig. 4 The third-order NLO of the pristine Al_4 precursor and compounds from supracluster assembly. (a) The opening Z-scan curve at 532 nm of the samples. At the top of the Z-scan curves are photographs of the Al₄ precursor@PDMS and AlOCs@PDMS before (upper panel) and after (lower panel) applying 360 nm UV light. (b) Curves of output fluence versus input fluence for the Al₄ precursor, and AlOC-168 to AlOC-176. (c) Plots of normalized transmittance versus input fluence for the Al₄ precursor, and AlOC-168 to AlOC-176. (d) Comparison of the nonlinear absorption coefficients at a laser intensity of 80 µJ for Al₄ precursor@PDMS and AlOCs@PDMS.

action significantly enhancing charge transfer between anion and cation groups, thus improving the third-order NLO performance. The supercluster assembly strategy may serve as a reference for future performance enhancement. It is a direction worth exploring further. It is comparable to that of rare-earth metal or transition metal cluster-based materials, ^{55,56} although they are not comparable with metal-porphyrinic frameworks or porphyrin covalent organic frameworks (Table S5†). ^{57,58}

The curves of output fluence *versus* input fluence of such samples show that the output fluence linearly increases at low-incident fluence. At high-incident fluence, the output fluence deviates from linearity and shows a typical OL response (Fig. 4b). Their optical limiting threshold values ($F_{\rm OL}$, defined as the input fluence at 50% linear transmittance) are 5.12 J cm⁻² ($F_{\rm OL}$ Ald precursor), 1.09 J cm⁻² ($F_{\rm OL}$ AloC-168), 3.45 J cm⁻² ($F_{\rm OL}$ AloC-169), 1.23 J cm⁻² ($F_{\rm OL}$ AloC-170(R)), 3.59 J cm⁻² ($F_{\rm OL}$ AloC-171), 0.61 J cm⁻² ($F_{\rm OL}$ AloC-172), 0.39 J cm⁻² ($F_{\rm OL}$ AloC-173), 0.86 J cm⁻² ($F_{\rm OL}$ AloC-174), 0.75 J cm⁻² ($F_{\rm OL}$ AloC-175), and 0.33 J cm⁻² ($F_{\rm OL}$ AloC-176), respectively (Fig. 4c). In addition, we have carried out cycling experiments to verify the laser stability of **AloCs@PDMS** materials. The results show that the RSA profiles of all third-order NLO materials have the same reproducibility after 4 or 8 cycles (Fig. S63–S72†).

Conclusions

In summary, by employing the Al $_4$ cluster as a precursor, a supracluster assembly strategy (including both non-covalent interaction and coordination bonds) was applied to construct a series of interesting cluster-based architectures. By introducing large π -conjugated phenanthroline- or phenylphosphine-type organic ligands, these cluster-assembled compounds showed different photoluminescent colors. The results showed that supracluster assembly via coordination bonds facilitates charge transfer and can effectively enhance the third-order NLO properties of cluster-based materials compared with those based on non-covalent bonds. This work not only demonstrates a general synthesis strategy based on a pre-designed precursor but also provides important guidance for stepwise tuning and optimization toward specific properties.

Experimental section

Synthesis of the Al₄ precursor

The synthesis of $(Me_2NH_2)_4[Al_4(L)_4(Cat)_2]$ (AlOC-13) is based on our previous work. A mixture of $Al(O^iPr)_3$ (0.2 g, 0.98 mmol), 2,3-dihydroxybenzoic acid $(H_3L, 0.4$ g, 2.60 mmol), pyrazole (2 g, 29.38 mmol), and DMF (5 mL) was sealed in a 25 mL Teflon reactor and transferred to a preheated oven at 160 °C for 3 days. When cooled to room temperature, light brown crystals of **AlOC-13** were obtained (yield: 75% based on $Al(O^iPr)_3$). Amplification synthesis is also possible:

using a 25 mL Teflon reactor, 0.27 g of **AlOC-13** can be obtained, and using a 50 mL Teflon reactor, 0.68 g of **AlOC-13** can be obtained.

Supracluster assembly based on the Al4 precursor

By investigating the supracluster assembly behavior of the Al₄ cluster with metal ions (such as Ag Fe²⁺ and Cu²⁺) in the presence of chelated ligands (like N,N-chelated (phenanthrolinetype) or P,P-chelated (phenylphosphine-type) ligands) with diverse binding distances and angles, we successfully obtained a variety of cluster-based solids. A typical solvothermal synthesis of AlOC-168 $(Me_2NH_2)_3[Ag(1,3-bdpp)_2][Al_4(L)_4(Cat)_2]$. guests is described as follows: the Al4 precursor (55 mg, 0.05 mmol) was dissolved in 2 mL of H₂O at 100 °C for 6 h. A mixture of 2 mL of precursor, silver perchlorate (AgClO₄, 14 mg, 0.07 mmol), 1,3-bis(diphenylphosphino)propane (1,3bdpp, 45 mg, 0.11 mmol), EtOH (2 mL), and DMF (2 mL) was sealed in a 20 mL vial at 80 °C for 24 h. When cooled to room temperature, light brown crystals were obtained (yield: 75% based on the Al₄ precursor). The crystals were rinsed with EtOH and preserved in a sealed and dry environment. The synthetic procedure for other compounds is similar to that of Aloc-168 but with different ligands and solvents. For the detailed synthesis, please refer to the ESI.†

Fabrication of Al₄ precursor@PDMS and AlOCs@PDMS

Al₄ precursor@PDMS and **AloCs@PDMS** were prepared using Sylgard 184 (Dow Corning) by thoroughly mixing 10 parts base with 1 part curing agent. The coordinate crystals were mixed with the PDMS solution to form coordinate-dispersed PDMS suspension, and then the mixture suspension was added to a template and the template was placed in a vacuum oven at 60 °C for 6 hours. Finally, transparent and flexible Al₄ precursor@PDMS and **AloCs@PDMS** were obtained.

Z-Scan measurements

The third-order NLO properties of the sample were evaluated by using the *Z*-scan technique. The excitation light source was an Nd:YAG laser with a repetition rate of 5 Hz. The laser pulse (period, 5 ns; wavelength, 532 nm) was split into two beams with a mirror. The pulse energies at the front and back of the samples were monitored using energy detectors 1 and 2. All of the measurements were conducted at room temperature. The sample was mounted on a computer-controlled translation stage that shifted each sample along the *z*-axis.

Author contributions

All authors contributed extensively to the work presented in this paper. J. Zhang and W.-H. Fang conceived the research project. F. Yang and R.-Y. Chen performed the synthesis, and characterization and optical property studies. T.-S. Wang and Y.-P. He helped with the crystallographic data refinement. W.-H. Fang and Y.-P. He wrote the manuscript and ESI† with input from the other authors.

Data availability

Research Article

X-ray crystallographic data have been deposited at the Cambridge Crystallographic Data Centre (https://www.ccdc.cam.ac.uk/) with CCDC reference numbers 2303391 to 2303400† (single-crystal AlOC-168 to AlOC-176).

Conflicts of interest

There are no conflicts to declare.

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Research Article

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