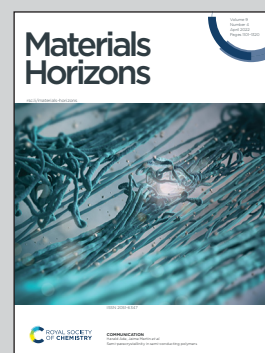


Showcasing research from Professor Dimos Poulikakos' laboratory, LTNT at ETH Zurich, Switzerland.

Microscale investigation on interfacial slippage and detachment of ice from soft materials

Surface icing is detrimental to applications ranging from transportation to biological systems. Soft elastomeric coatings can engender remarkably low ice adhesion strength. Here we present a methodology based on confocal traction force microscopy to investigate *in situ* ice adhesion on soft elastomeric materials at the microscale. We report unprecedented findings of the in-plane transient heterogeneous deformation field of the elastomer during the ice removal. Our findings provide real time insight into the complex process of rate-dependent microscopic deformation in viscoelastic solids during adhesion in general.

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Surface icing is detrimental to applications ranging from transportation to biological systems. Soft elastomeric coatings can engender remarkably low ice adhesion strength, but mechanisms at the microscale and resulting ice extraction outcomes need to be understood. Here we investigate dynamic ice-elastomer interfacial events and show that the ice adhesion strength can actually vary by orders of magnitude due to the shear velocity. We study the detailed deformation fields of the elastomer using confocal traction force microscopy and elucidate the underlying mechanism. The elastomer initially undergoes elastic deformation having a shear velocity dependent threshold, followed by partial relaxation at the onset of slip, where velocity dependent “stick-slip” micropulsations are observed. The results of the work provide important information for the design of soft surfaces with respect to removal of ice, and utility to fields exemplified by adhesion, contact mechanics, and biofouling.

Introduction

Soft viscoelastic solids (or simply elastomers) exhibiting both elastic and viscous behavior are ubiquitous in nature and practical applications. Biological tissues mainly comprise viscoelastic components with each tissue having a specific function.¹ From a practical point of view, elastomers are traditionally used in foams, adhesives, food additives, and so on.² With recent technological advances, the applications

Microscale investigation on interfacial slippage and detachment of ice from soft materials[†]

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New concepts

We present a methodology based on confocal traction force microscopy to investigate *in situ* ice adhesion on soft elastomeric materials at the microscale. We report unprecedented findings of the in-plane transient heterogeneous deformation field of the elastomer during the ice removal. We study the response of the elastomer at different ice removal rates, and directly relate the determined increase in ice adhesion stress to the applied shear rate. Further, we directly observe and visualize the process of interfacial slippage and subsequent relaxation of a soft surface, including the manifestation of an unexpected stick-slip phenomenon, which we explain. Our findings provide real time insight into the complex process of rate-dependent microscopic deformation in viscoelastic solids during adhesion in general, which is of significance in areas exemplified by adhesion and bonding materials, contact mechanics, bio-fouling control and soft robotic materials.

extend to the exciting domains of soft robotics, tissue engineering, flexible electronics, and so on.^{3–5} One of the potentially important applications of elastomers is their passive icephobic performance during harsh environmental icing conditions.^{6–9} Ice adhered to surfaces incurs malfunctioning or a drastic decrease in efficiency in many industrial settings.¹⁰ It is previously shown that elastomers offer exceptionally low ice adhesion shear strength, $\tau_{\text{ice}} = F^*/A \leq 10$ kPa, either through interfacial fracture- or slippage mechanisms.^{6,8,9,11–13} Here, F^* indicates the peak ice removal force recorded by a force gauge in a standard ice adhesion shear test at a constant shear velocity, V , and A is the apparent substrate-ice interfacial area.¹⁴ In contrast, rigid and liquid impregnated surfaces (which have an intervening lubricant layer in between ice and the base solid) have a lower limit of $\tau_{\text{ice}} \approx 50$ and 15 kPa, respectively.^{15–20} On this class of materials, τ_{ice} can increase significantly due to the highly humid ambient conditions or lubricant depletion over time.^{21–29} However, elastomers do not have such bottlenecks and are shown to be durable over several (10–20) icing/de-icing cycles.^{6,8} Other strategies with weakly interacting surface layers using organogel materials to mitigate icing on surfaces are discussed in ref. 30.

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Interfacial fracture on elastomers can be either stress- or toughness-limited de-bonding of ice, where the elastomer cavitation leads to easy ice removal.^{6,8} Slippage occurs when the elastomer chains adhered to the ice are sufficiently mobile, which can be realized by either reducing the cross-linking density of the elastomer or infiltrating the elastomer network with a lubricant.^{6,7,31,32} The latter is unlike liquid impregnated surfaces, because the lubricant is introduced into the bulk of the elastomer, and any lubricant layer on the interface interacting with ice is avoided. During slippage, ice remains adhered to the elastomer but continues to slide away from its initial location as long as shear is applied. Whether ice fractures or slips during an adhesion test is governed by a multitude of parameters: elastomer surface temperature, T , shear velocity, V , type of stress applied in the adhesion test (shear, mixed, and normal), elastomer thickness, h , and Young's modulus, E (also a function of temperature). This illustrates the complexity involved in comprehending the ice removal mechanisms on elastomers across a much broader spectrum of influencing conditions.

The ice adhesion shear strength, τ_{ice} , is the accepted defining metric for the performance of a surface, with respect to its resistance to ice detachment. In previous studies, τ_{ice} on elastomers is typically reported for low shear velocities of $V \leq 0.1 \text{ mm s}^{-1}$.^{6,8,13} Here, we show that τ_{ice} on soft silicone elastomeric coatings ($E = 15.4 \text{ kPa}$ at $T = -20 \text{ }^\circ\text{C}$) can increase by an order of magnitude, depending on the shear velocity, V , which can potentially define the domain of their applicability in practical settings. To understand the reason for this behavior, we first probe into the elastomer deformation using confocal traction force microscopy (cTFM) providing the necessary high-resolution visualization of spatio-temporal in-plane deformation of the elastomer, during shear mode experiments. Therefore, the cTFM technique provides unprecedented detail of the elastomer deformation at the microscopic scale. We then invoke stochastic models involving cyclic adsorption and desorption of elastomer chains near the interface to explain the observed trend of τ_{ice} with V . Further, we investigate the effect of mode of adhesion test (shear, mixed, and normal) and elastomer temperature on the ice adhesion strength, and show that the trend of τ_{ice} with V in mixed mode tests is similar to that of shear mode tests while τ_{ice} decreases with T .

Experimental section

Macroscopic ice adhesion setup

For the macroscopic ice adhesion experiments, we follow a similar protocol as explained in ref. 14. A cryostage (Linkam BCS196) with liquid Nitrogen supply is operated with a temperature controller unit (Linkam TMS93). A cylindrical copper block with a circular groove (diameter 25 mm, depth 0.17 mm) on the top is clamped firmly on the cryostage. A double-sided tape (TESA Doppelband-Fotostrip) is used to fix the spin coated circular cover slips inside the groove. In addition, Kapton tape (3M Polyimide Tape 8997) is also used around the edge on the

top side of the surface to make sure that the substrate is firmly held in its position during the adhesion test. A cylindrical glass cuvette (outer diameter 8.3 mm, inner diameter 6.5 mm, height 10 mm) which is open on both sides is gently placed on the substrate. The cross-section of the cuvette which comes in contact with the elastomer is polished using wet sand papers with 800, 1200, and 2000 grit in that order to ensure surface flatness. The cuvette is filled with $\approx 0.25 \text{ mL}$ of DI water using a syringe by hand. The experiments are performed in Nitrogen ambient to reduce the relative humidity, RH, so that frost formation on the sample is avoided. An acrylic chamber with slots to accommodate force pin from the top and the side is mounted on the cryostage to ensure the Nitrogen ambient. The slot on the top is sealed while conducting shear or mixed mode adhesion tests. Likewise, the slot on the side is sealed while conducting normal mode tests. The relative humidity, RH, and ambient temperature, $T_{amb} = 21 \text{ }^\circ\text{C}$, in the chamber are measured using a sensor (Sensirion Ag SHT31 Smart gadget). Perfect sealing is not a critical aspect of the experiment; in all the macroscopic experiments, $\text{RH} < 5\%$ even with one slot being fully open.

For measuring the force during the adhesion tests, either a Mark 10 force gauge M5-5 (calibration certified) or Tekscan A201 force sensor 0–4 N (calibrated in-house) are used. The force pin comes in contact with the wall of the cuvette at a height of $< 1 \text{ mm}$ and 6.1 mm for the shear and mixed mode tests, respectively. In order to ensure that the contact area of the force pin with the cuvette is minimum, a taper is provided near its end so that the contact width of the pin with the cuvette is $\approx 85 \text{ } \mu\text{m}$. The force gauge is mounted on top of a manual stage (Thorlabs XR25C) whose axis in the horizontal plane is oriented orthogonal to the shear direction to precisely control the gauge position. This stage is assembled on a lab jack (Thorlabs L490) to control the vertical position of the force gauge. The entire assembly is mounted on a motorized stage (Thorlabs NRT100/M) which is operated at a desired velocity (0.01, 0.1, 1, and 10 mm s^{-1}) for the experiments. The temperature of the elastomer surface is measured using a surface mounted T-type thermocouple (Omega, 5SRTC-TT-TI-40-1M) connected to a thermocouple reader (Lutron TM947SD). The experimental range of elastomer temperature, T , is -33 to $-9 \text{ }^\circ\text{C}$. The ice adhesion test is also captured optically from the side view using a digital microscope (Celestron Digital Microscope Pro, 5 MP) operated at 9–12 Hz for $V \leq 1 \text{ mm s}^{-1}$. For $V = 10 \text{ mm s}^{-1}$, a high-speed camera (Photron SA 1.1) operated at 1000 Hz is used.

The acquisition rate of force gauge is set at 20, 20, 50, and 1000 Hz when the velocities are 0.01, 0.1, 1, and 10 mm s^{-1} , respectively. From the force, F , vs. time, t , plots during the ice adhesion test, the maximum force, F^* , is extracted to estimate ice adhesion shear and normal strengths, τ_{ice} (or σ_{ice}) = F^*/A where $A = \pi R^2$ is the cross-sectional area of the ice in contact with the elastomer. The estimation of σ_{ice} in the mixed mode is discussed in ref. 33. The adhesion strengths did not vary significantly with respect to icing/de-icing cycles on a given sample.



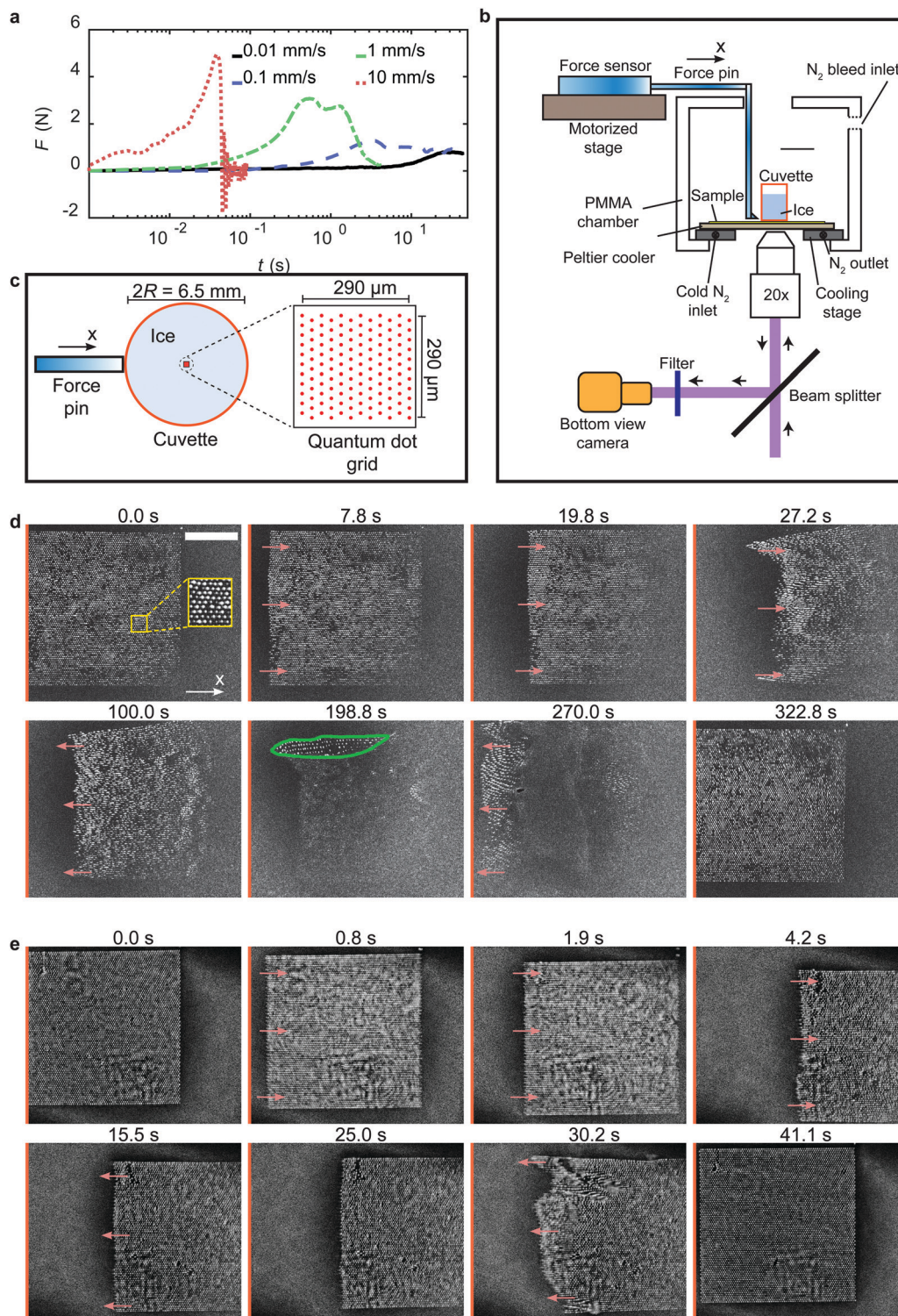


Fig. 1 Icephobicity of elastomers depends on shear velocity. (a) Representative force, F , vs. time, t , plots for an ice adhesion test in shear mode at different shear velocity, V , with elastomer thickness, $h = 35 \mu\text{m}$, and surface temperature, $T = -20 \text{ }^\circ\text{C}$. The peak ice removal force, F^* , increases with V . Time, $t = 0$ is taken just before the instance when the force has non-zero values. (b) Sketch (not to scale) showing the experimental setup of cTFM. (c) Sketch (not to scale) showing the elastomer printed with fluorescent quantum dots (QDs) grid in red with each dot having a diameter of $\approx 0.2 \mu\text{m}$. The cuvette inner diameter, $2R = 6.5 \text{ mm}$, is much larger than the grid dimension ($290 \times 290 \mu\text{m}$). As ice is sheared, the deformation of the QDs is captured from the bottom view. (d and e) Selected bottom view snapshots of the QD grid (and with it the elastomer surface) during the shear mode test at $V = 0.01$ and 0.1 mm s^{-1} , respectively, with $h = 35 \mu\text{m}$, and $T = -20 \text{ }^\circ\text{C}$. Time, $t = 0$ in (d and e) represent the reference configuration at the instance just before the QDs start to translate. The orange line indicates the reference axis as a guide to the eye and the inset in yellow represents a zoomed in portion of the QD grid. The red arrows indicate the direction of motion of the QD grid. At $t = 198.8 \text{ s}$, the region enveloped by the green line indicates the air pocket. Scale bar: (d), $100 \mu\text{m}$ (same in (e)).



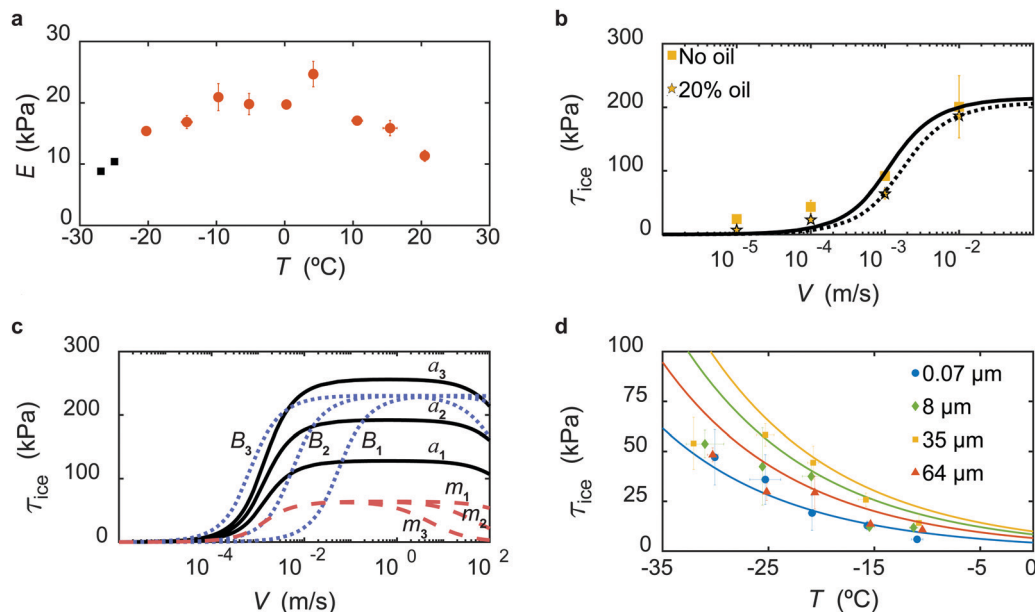


Fig. 3 Effect of temperature and shear velocity on ice adhesion in shear mode test. (a) Non-monotonic behaviour of Young's modulus of the elastomer, E , with surface temperature, T . (b) Variation of τ_{ice} with shear velocity, V , at an elastomer (without and with 20 wt% Silicon oil) surface temperature, $T = -20$ °C, and thickness, $h = 35$ μm . The solid and dashed black curves indicate the best fit of Chernyak and Leonov adhesive friction model (eqn (1)) to the experimental data without and with oil, respectively. (c) Variation of τ_{ice} with shear velocity for different Chernyak and Leonov parameters for arbitrarily selected values of a , B , and m . When only a increases ($a_1 < a_2 < a_3$), the peak value of τ_{ice} increases but the velocity at which the peak is obtained remains constant (black solid lines). When only B increases ($B_1 < B_2 < B_3$), the curve shifts to the left (blue dashed lines) but the peak value of τ_{ice} remains the same. When only m increases ($m_1 < m_2 < m_3$), the plateau width after attaining the peak value of τ_{ice} reduces (red dashed lines). (d) Variation of τ_{ice} with elastomer surface temperature, T , at $V = 0.1$ mm s^{-1} . The solid curves indicates the Williams–Landel–Ferry transformation to the experimental data for different elastomer thickness, h . Error bars represent standard deviation for $n_e \geq 3$ independent experiments in (a, b and d). The black data points in (a) represent individual data points.

The variation of τ_{ice} with V for $h = 35$ μm , at $T = -20$ °C is shown in Fig. 3b. Assuming $\tau_{\text{friction}} = \tau_{\text{ice}}$ with $G = 5.13$ kPa at $T = -20$ °C, we obtain the parameters through non-linear curve fitting, $a = 83.83$, $B = 241$ s m^{-1} , and $m = 2.3 \times 10^{-8}$, all of which seem quite reasonable.⁴⁸ We remark here that the viscous retardation in the chains is neglected in our study; due consideration of this phenomenon would result in $\tau'_{\text{friction}} = (1 + s)\tau_{\text{friction}}$. Here, s represents the ratio of viscous retardation time of the chain with the average bound time in the rest state, t_{0b} . Since $s \leq m \leq 1$,⁴⁸ we have $\tau'_{\text{friction}} \approx \tau_{\text{friction}}$ in our case as $m \ll 1$. However, as $s \rightarrow 1$, *i.e.* when the viscoelastic effects of each polymer chain are significant, the retardation effects can become significant.

The parameter m correlates with the width of the plateau; increasing m reduces the plateau width. Since a correlates with the maximum stretch of the elastomer chain at the instance of detaching, relatively mobile chains will have lower magnitude of a (see Fig. 3c). Reducing the parameter B ($1/B$ represents a molecular velocity scale) merely shifts the curve to right. Finally, $\tau_{\text{ice}} \propto G$ as described in previous works.^{6,7} Although, controlling the parameters a , m , and B , is challenging experimentally, the insight their variation provides is valuable. The shear modulus, G , we can easily modify and observe the trend of τ_{ice} with V . The infusion of oil into the elastomer reduces the cross-linking density (thereby reducing shear modulus, G),⁶ which leads to a reduction of τ_{ice} . As an example, with all the

experimental conditions being the same as in Fig. 3b, we show that infusing the polymer network with 20% silicone oil leads to a reduction in τ_{ice} (dashed line in Fig. 3b) with $a = 117.7$, $B = 159.9$ s m^{-1} , and $m = 1.4 \times 10^{-7}$. The modified shear modulus when oil is infused is estimated as $(1 - \omega)^{5/3}G = 3.54$ kPa where ω corresponds to the oil weight fraction in the polymer network.⁷ This indicates that the shear modulus (reduced by 30% in comparison to no oil case) has a dominant effect on τ_{ice} compared to the interfacial chain mobility (a increased by 40% in comparison to the no oil case). This is also supported from the mathematical form of τ_{friction} (eqn (1)). The areal density of active load bearing chains $N \propto G$,⁷ and therefore we conclude that the areal density of chains has a dominant effect than the interfacial mobility of chains. Finally, the curve shifts to the right since B reduced while m increased which is also a desired consequence from oil addition as the plateau width reduces.

We then investigate the effect of elastomer surface temperature, T , on τ_{ice} at $V = 0.1$ mm s^{-1} , for $h = 35$ μm (see Fig. 3d) where τ_{ice} decreases with T . Employing an analogy from the variation of elastomer friction with temperature, we use the classic empirical Williams–Landel–Ferry (WLF) transformation to explain these results.^{50,51} The transformation indicates that the temperature dependence of mechanical processes on elastomers can be estimated using a single empirical function. Accordingly, we have $\log(\tau_{\text{ice}}/\tau_{\text{ref}}) = [-Y(T - T_{\text{ref}})]/[Z + T - T_{\text{ref}}]^{-1}$ where $\tau_{\text{ref}} = \tau_{\text{ice}}$ at $T_{\text{ref}} = 253$ K, and $Y = 30$ and $Z = 900$ K are



parameters a , B , and m being the same as in Fig. 3b (no oil case). The pre-factor of $\tau_{\text{ice},s}$ varies slightly in the experiments but is of the same order of the ratio l_s/l_m and more importantly, $\tau_{\text{ice},m}$ and $\tau_{\text{ice},s}$ are directly proportional at all values of V . This remarkably simple proposition also works well for different elastomer thicknesses, h (see Fig. S6, ESI†). By definition, the parameters a , B , and m should not vary in the mixed mode test. We conjecture that the presence of elastic instabilities in the mixed mode test would reduce the number of active load bearing chains, N , as compared to the shear mode test, which leads to a reduction of τ_{ice} . Fig. 5b shows the variation of ice adhesion strength in the normal mode test, σ_{ice} , with V for $h = 35 \mu\text{m}$ at $T = -20 \text{ }^\circ\text{C}$. In this case, V represents the pull up velocity of the force gauge. σ_{ice} increases with V in a similar fashion as in shear and mixed mode tests. However, the error is significantly higher (see also Fig. S7, ESI†) making it difficult to determine a clear dependency with V . We attribute this behavior to the partial cohesive failure of the elastomer during the adhesion test (see Fig. S8, ESI†).

Fig. 5c and d show the variation of ice adhesion strength (τ_{ice} , σ_{ice} in mixed mode and σ_{ice} in normal mode) with T at $V = 0.1 \text{ mm s}^{-1}$. Again, using WLF transformation in the mixed mode, we can estimate the temperature dependence of τ_{ice} with T (see Fig. 5c and Fig. S9, ESI†). σ_{ice} in the normal mode remains constant in the entire temperature domain for a given h . Further, the magnitude of σ_{ice} in the normal mode (see Fig. 5d) is much higher than that in the mixed mode. Again, the error is significantly high to predict any conclusive trend (see also Fig. S10, ESI†).

Conclusion

In summary, we show that the velocity at which the force is imposed on ice during ice removal, is an important factor significantly affecting the ice adhesion strength on elastomers. By employing cTFM, we quantified the transient non-uniform elastic deformation fields of the elastomer and related it to the applied external shear stress. The maximum deformation increases significantly with shear velocity, and consequently τ_{ice} increases with V . Further, we observe pulsations in the slip regime at high velocity, which agrees well with the reported observations in the literature. The capability to visualize the deformation of the elastomer at high resolution is expected to be useful in studies exemplified by bio-fouling control, viscoelastic adhesives, contact mechanics, and soft robotics.

Author contributions

T. M. S. and D. P. designed research; K. R. and J. G., performed research and analyzed the data; all the authors contributed in writing the paper.

Conflicts of interest

D. P. is participating in Scrona, an ETH spin-off, which is developing the electrohydrodynamic NanoDrip-printing technology

towards commercialization. The remaining authors declare no competing interests.

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References

- 1 D. Huang, Y. Huang, Y. Xiao, X. Yang, H. Lin, G. Feng, X. Zhu and X. Zhang, *Acta Biomater.*, 2019, **97**, 74–92.
- 2 R. S. Lakes, *Viscoelastic Solids*, CRC Press, 2017, pp. 1–476.
- 3 C. M. Tringides, N. Vachicouras, I. de Lázaro, H. Wang, A. Trouillet, B. R. Seo, A. Elosegui-Artola, F. Fallegger, Y. Shin, C. Casiraghi, K. Kostarelos, S. P. Lacour and D. J. Mooney, *Nat. Nanotechnol.*, 2021, **16**, 1019–1029.
- 4 S. I. Rich, R. J. Wood and C. Majidi, *Nat. Electron.*, 2018, **1**, 102–112.
- 5 M. Cianchetti, C. Laschi, A. Menciassi and P. Dario, *Nat. Rev. Mater.*, 2018, **3**, 143–153.
- 6 K. Golovin, S. P. R. Kobaku, D. H. Lee, E. T. DiLoreto, J. M. Mabry and A. Tuteja, *Sci. Adv.*, 2016, **2**, e1501496.
- 7 K. Golovin and A. Tuteja, *Sci. Adv.*, 2017, **3**, e1701617.
- 8 K. Golovin, A. Dhyani, M. D. Thouless and A. Tuteja, *Science*, 2019, **364**, 371–375.
- 9 D. L. Beemer, W. Wang and A. K. Kota, *J. Mater. Chem. A*, 2016, **4**, 18253–18258.
- 10 M. J. Kreder, J. Alvarenga, P. Kim and J. Aizenberg, *Nat. Rev. Mater.*, 2016, **1**, 15003.
- 11 M. K. Chaudhury and K. H. Kim, *Eur. Phys. J. E: Soft Matter Biol. Phys.*, 2007, **23**, 175–183.
- 12 J. Y. Chung and M. K. Chaudhury, *J. Adhes.*, 2005, **81**, 1119–1145.
- 13 P. Iradjizad, A. Al-Bayati, B. Eslami, T. Shafquat, M. Nazari, P. Jafari, V. Kashyap, A. Masoudi, D. Araya and H. Ghasemi, *Mater. Horiz.*, 2019, **6**, 758–766.
- 14 A. J. Meuler, J. D. Smith, K. K. Varanasi, J. M. Mabry, G. H. McKinley and R. E. Cohen, *ACS Appl. Mater. Interfaces*, 2010, **2**, 3100–3110.
- 15 P. Kim, T. S. Wong, J. Alvarenga, M. J. Kreder, W. E. Adorno-Martinez and J. Aizenberg, *ACS Nano*, 2012, **6**, 6569–6577.
- 16 A. J. Meuler, G. H. McKinley and R. E. Cohen, *ACS Nano*, 2010, **4**, 7048–7052.
- 17 A. Davis, Y. H. Yeong, A. Steele, I. S. Bayer and E. Loth, *ACS Appl. Mater. Interfaces*, 2014, **6**, 9272–9279.
- 18 Y. H. Yeong, A. Millionis, E. Loth, J. Sokhey and A. Lambourne, *Langmuir*, 2015, **31**, 13107–13116.
- 19 R. Dou, J. Chen, Y. Zhang, X. Wang, D. Cui, Y. Song, L. Jiang and J. Wang, *ACS Appl. Mater. Interfaces*, 2014, **6**, 6998–7003.
- 20 J. Lv, Y. Song, L. Jiang and J. Wang, *ACS Nano*, 2014, **8**, 3152–3169.



- 21 K. K. Varanasi, T. Deng, J. D. Smith, M. Hsu and N. Bhate, *Appl. Phys. Lett.*, 2010, **97**, 234102.
- 22 S. B. Subramanyam, K. Rykaczewski and K. K. Varanasi, *Langmuir*, 2013, **29**, 13414–13418.
- 23 L. Hauer, W. S. Y. Wong, A. Sharifi-Aghili, L. Kondic and D. Vollmer, *Phys. Rev. E*, 2021, **104**, 044901.
- 24 X. Liu, H. Zhao, P. Li, Y. Pang, Y. Fan, B. Zhang, L. Wang, Y. Zheng and Z. Wang, *ChemNanoMat*, 2019, **5**, 175–180.
- 25 Y. Hou, M. Yu, Y. Shang, P. Zhou, R. Song, X. Xu, X. Chen, Z. Wang and S. Yao, *Phys. Rev. Lett.*, 2018, **120**, 75902.
- 26 P. Papadopoulos, L. Mammen, X. Deng, D. Vollmer and H.-J. Butt, *Proc. Natl. Acad. Sci. U. S. A.*, 2013, **110**, 3254–3258.
- 27 J. Liu, M. Kappl and H. J. Butt, *Matter*, 2020, **3**, 981–983.
- 28 Y. Hou, H. Butt and M. Kappl, *Ice Adhes.*, 2020, 55–85.
- 29 J. Chen, J. Liu, M. He, K. Li, D. Cui, Q. Zhang, X. Zeng, Y. Zhang, J. Wang and Y. Song, *Appl. Phys. Lett.*, 2012, **101**, 111603.
- 30 J. Lv, X. Yao, Y. Zheng, J. Wang and L. Jiang, *Adv. Mater.*, 2017, **29**, 1703032.
- 31 K. Vorvolakos and M. K. Chaudhury, *Langmuir*, 2003, **19**, 6778–6787.
- 32 B. Z. Newby, M. K. Chaudhury and H. R. Brown, *Science*, 1995, **269**, 1407–1409.
- 33 T. Maitra, S. Jung, M. E. Giger, V. Kandrical, T. Ruesch and D. Poulidakos, *Adv. Mater. Interfaces*, 2015, **2**, 1500330.
- 34 J. Gerber, T. Lendenmann, H. Eghlidi and T. M. Schutzius, *Nat. Commun.*, 2019, **10**, 4776.
- 35 L. B. Boinovich, K. A. Emelyanenko and A. M. Emelyanenko, *J. Colloid Interface Sci.*, 2022, **606**, 556–566.
- 36 M. Bergert, T. Lendenmann, M. Zündel, A. E. Ehret, D. Panozzo, P. Richner, D. K. Kim, S. J. P. Kress, D. J. Norris, O. Sorkine-Hornung, E. Mazza, D. Poulidakos and A. Ferrari, *Nat. Commun.*, 2016, **7**, 12814.
- 37 P. Galliker, J. Schneider, H. Eghlidi, S. Kress, V. Sandoghdar and D. Poulidakos, *Nat. Commun.*, 2012, **3**, 1891.
- 38 S. J. P. Kress, P. Richner, S. V. Jayanti, P. Galliker, D. K. Kim, D. Poulidakos and D. J. Norris, *Nano Lett.*, 2014, **14**, 5827–5833.
- 39 T. Lendenmann, T. Schneider, J. Dumas, M. Tarini, C. Giampietro, A. Bajpai, W. Chen, J. Gerber, D. Poulidakos, A. Ferrari and D. Panozzo, *Nano Lett.*, 2019, **19**, 6742–6750.
- 40 A. E. Carte, *Proc. Phys. Soc., London*, 1961, **77**, 757–768.
- 41 V. Mhetar and L. A. Archer, *Macromolecules*, 1998, **31**, 8617–8622.
- 42 R. Hopf, L. Bernardi, J. Menze, M. Zündel, E. Mazza and A. E. Ehret, *J. Mech. Behav. Biomed. Mater.*, 2016, **60**, 425–437.
- 43 B. Pan, K. Qian, H. Xie and A. Asundi, *Meas. Sci. Technol.*, 2009, **20**, 062001.
- 44 R. W. Style, R. Boltyskiy, Y. Che, J. S. Wettlaufer, L. A. Wilen and E. R. Dufresne, *Phys. Rev. Lett.*, 2013, **110**, 066103.
- 45 R. W. Style, Y. Che, S. J. Park, B. M. Weon, J. H. Je, C. Hyland, G. K. German, M. P. Power, L. A. Wilen, J. S. Wettlaufer and E. R. Dufresne, *Proc. Natl. Acad. Sci. U. S. A.*, 2013, **110**, 12541–12544.
- 46 M. K. Chaudhury, K. Vorvolakos and D. Malotky, arXiv:1508.01452, 2015, 1–15.
- 47 A. Schallamach, *Wear*, 1963, **6**, 375–382.
- 48 Y. B. Chernyak and A. I. Leonov, *Wear*, 1986, **108**, 105–138.
- 49 M. H. Sadd, *Elasticity: Theory, applications, and numerics*, 3rd edn, 2014.
- 50 M. L. Williams, R. F. Landel and J. D. Ferry, *J. Am. Chem. Soc.*, 1955, **77**, 3701–3707.
- 51 W. Gnorich and K. A. Grosch, *Rubber Chem. Technol.*, 1975, **48**, 527–537.

