

PAPER

[View Article Online](#)
[View Journal](#) | [View Issue](#)
Cite this: *Nanoscale*, 2025, **17**, 1990

Sulfide-mediated growth of NIR luminescent Pd/Ag atomically precise nanoclusters†

 Yu-Rong Ni,^a Michael N. Pillay,^{ID} ^a Tzu-Hao Chiu,^a Hao Liang,^b Samia Kahlal,^b Jie-Ying Chen,^c Yuan-Jang Chen,^c Jean-Yves Saillard ^{ID} ^{*b} and C. W. Liu ^{ID} ^{*a}

An essential feature of coinage metal nanoclusters (NCs) is their photoluminescence (PL), which spans a wide range of wavelengths from visible to near-infrared regions (NIR-I/II). A key challenge for synthetic chemists is to develop materials capable of efficient spectral change with maximum efficiency. Herein, we report novel dithiolate-protected bimetallic Pd-Ag NCs of the type $[\text{PdAg}_{16}\text{S}_2\{\text{S}_2\text{P}(\text{OR})_2\}_{12}]$ ($\text{R} = {}^i\text{Pr}$, **1_{Pr}** and **1_{Bu}**) and $[\text{Pd}_6\text{Ag}_{14}\text{S}\{\text{S}_2\text{P}(\text{O}^i\text{Bu})_2\}_{12}]$ (**2_{Bu}**). Sulfide-mediated expansions of NCs result in unique PL in the NIR-I region for **1_{Pr}** and **1_{Bu}** ($\lambda_{\text{max}} = 808$ and 811 nm) and the NIR-II region for **2_{Pr}** ($\lambda_{\text{max}} = 1007$) at 77 K. NIR PL enhancement largely depends on structural modification with the sulfide anions at the central position. DFT calculations indicate that the PL properties are associated with $4d_x(\text{Pd})/3p_x(\text{S}) \rightarrow 5s/5p(\text{Ag})$ excitation, resulting from the existence of S–Pd(0)–S motifs in both **1** and **2**. The electrochemical gaps of **1_{Pr}**, **1_{Bu}**, and **2_{Bu}** are recorded by SWV.

 Received 8th October 2024,
 Accepted 3rd December 2024

DOI: 10.1039/d4nr04136d

rsc.li/nanoscale

Introduction

Atomically precise nanoclusters (NCs) have emerged as a material of the future, offering unique advantages over their bulk counterparts. One key improvement is the ability to modulate the energy levels and tuning energy band gaps, thus influencing both optical and electrochemical performances. Photoluminescence (PL) in the near-infrared (NIR) region is highly desirable due to its minimal interaction with biological tissues, making it a safe and effective tool for various applications in the biomedical field,^{1,2} such as cell labeling,³ bio-sensing,⁴ and bioimaging.^{5–7}

Over the past decade, substantial progress has been made in synthesizing and characterizing coinage metal NCs with enhanced PL quantum yields (PLQYs). However, achieving PL in the NIR-II region (1000–1700 nm) remains challenging. Coinage metal NCs have shown some promise, with a few examples of gold,^{8–15} silver¹⁶ and copper¹⁷ reported. While noble metal-doped or alloyed Ag NCs can extend emission from the UV/vis to the NIR-I range,^{18–31,38–40} achieving longer

wavelengths (1000–1700 nm) remains elusive. To achieve the NIR-II photoluminescence of silver NCs, we propose in this report the addition of a sulfur heteroatom into the metallic framework to induce emission at longer wavelengths. Various NC morphologies have been isolated due to the versatility of sulfide, which can occupy multiple positions within the NC, not just the central one.^{32,33} In several cases, the inclusion of sulfide in the NC primary sphere directs the growth of the subsequent layers and plays a crucial role in the assembly of large arrangements.^{34–37}

Herein, we report the synthesis, composition, and optical characteristics of the selective incorporation of sulfide anions into positions of the metallic primary layer, resulting in bimetallic Pd–Ag NCs $[\text{PdAg}_{16}\text{S}_2\{\text{S}_2\text{P}(\text{OR})_2\}_{12}]$ ($\text{R} = {}^i\text{Pr}$, **1_{Pr}** and **1_{Bu}**) and $[\text{Pd}_6\text{Ag}_{14}\text{S}\{\text{S}_2\text{P}(\text{O}^i\text{Bu})_2\}_{12}]$ (**2_{Bu}**). Single-crystal X-ray diffraction (SCXRD), X-ray photoelectron spectroscopy (XPS), ESI-MS, optical analysis, and multinuclear NMR spectroscopy determined the composition and structural properties and their electronic structures were analyzed through DFT calculations.

Results and discussion

Building upon our earlier report on sulfide-centered Pd/Ag of the type $[\text{Pd}_6\text{Ag}_{14}(\text{S})\{\text{S}_2\text{P}(\text{O}^n\text{Pr})_2\}_{12}]$ (**2_{Pr}**), which involves the cleavage of the P–S bond from the dithiophosphate and the *in situ* sulfide generation,¹⁹ we sought to develop a more general protocol for incorporating sulfide into NCs and evaluate its reproducibility. To achieve this, we varied the alkoxy

^aDepartment of Chemistry, National Dong Hwa University, Hualien 97401, Taiwan, Republic of China. E-mail: chenwei@gms.ndhu.edu.tw

^bUniv Rennes, CNRS, ISCR-UMR 6226, F-35000 Rennes, France.
E-mail: jean-yves.saillard@univ-rennes.fr

^cDepartment of Chemistry, Fu Jen Catholic University, New Taipei City, 24205 Taiwan, Republic of China

† Electronic supplementary information (ESI) available. CCDC 2379720–2379722. For ESI and crystallographic data in CIF or other electronic format see DOI: <https://doi.org/10.1039/d4nr04136d>

substituent on the dithiophosphate ligand (dtp), utilizing both isopropoxy and isobutoxy derivatives.

This strategy (Scheme 1) successfully yielded two novel disulfide nanoclusters, $[\text{PdAg}_{16}(\text{S})_2\{\text{S}_2\text{P}(\text{OR})_2\}_{12}]$ (**1_{Pr}**, $\text{R} = \text{iPr}$; **1_{Bu}**, $\text{R} = \text{iBu}$), in addition to the sulfide-centered clusters, $[\text{Pd}_6\text{Ag}_{14}(\text{S})\{\text{S}_2\text{P}(\text{O}^i\text{Bu})_2\}_{12}]$ (**2_{Bu}**), of which the $[\text{Pd}_6\text{Ag}_{14}(\text{S})\{\text{S}_2\text{P}(\text{O}^i\text{Pr})_2\}_{12}]$ (**2_{Pr}**) relative is known.¹⁹ The products are isolated and purified by column chromatography, affording **1** and **2** in moderate yields.

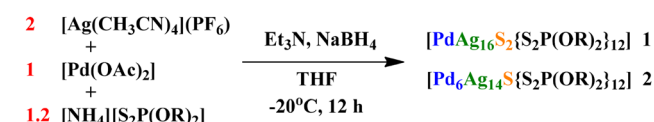
The purity and composition of the isolated NCs were confirmed by mass spectrometry, Fig. 1. The mass spectrum of **1_{Pr}** exhibits a dominant peak assigned to $[\text{1_{Pr}} + \text{Ag}]^+$ at m/z 4563.4390 Da (calc.: m/z 4563.4375 Da), Fig. 1a. The simulated peak is given in Fig. S1,[†] compared with a silver and proton adduct to demonstrate the presence of a palladium atom. The dominant m/z peak for **1_{Bu}** is assigned to the silver adduct $[\text{1_{Bu}} + \text{Ag}]^+$ at 4899.8438 Da (calc.: m/z 4899.8140 Da), Fig. 1b and S2.[†] Similarly, cluster **2_{Bu}** forms a silver adduct of the type $[\text{2_{Bu}} + \text{Ag}]^+$ at m/z 5184.6678 Da (calc.: m/z 5184.5506 Da), Fig. 1c. The experimental and simulated isotopic distributions

exhibit a strong correlation, further supporting the assigned compositions. Additionally, XPS analysis confirms the presence of Pd(0) and Ag(I) in the NCs, as evidenced by their characteristic binding energies, Fig. S3–5.[†] The temperature-dependent UV-vis spectra are shown in Fig. S13–15[†] and they were used to investigate the stability of the NCs.

Structural analysis

The atomic arrangements of **1_{Pr}** and **1_{Bu}** were determined by SCXRD analysis and their refinement details are given in Table S1.[†] Selected interatomic distances are given in Table 1. **1_{Pr}** and **1_{Bu}** crystallize in two different space groups, with **1_{Pr}** being solvated by a dichloromethane molecule. Otherwise, they are isostructural (apart from small differences in the ligand coordination modes, which are discussed later). Thus, the following discussion is based on **1_{Bu}** (see Fig. 2). Its highly unique core structure features a Pd-centered Ag_{10} pentagonal antiprism kernel, with two sulfides capping each of the pentagonal faces (Fig. 2a). It can thus be seen also as a Pd-centered Ag_{10}S_2 icosahedron with approximate D_{5d} symmetry and can be compared with typical species having Pd@Ag_{12} icosahedral cores.²¹ Substituting two vertices with sulfur atoms results in a large contraction compared to the regular Ag_{12} icosahedron, with $\text{Pd}\cdots\text{S}$ distances (ca. 2.30 Å) much shorter than the $\text{Pd}\cdots\text{Ag}$ in Pd@Ag_{12} species (ca. 2.90 Å).

Furthermore, the differential bonding between Pd–S and Ag–S is crucial for stabilizing this unique kernel configuration, which is unlikely to form in the absence of the central Pd



Scheme 1 Strategic incorporation of sulfide into bimetallic NCs.



Fig. 1 ESI-TOF-MS of spectra for **1_{Pr}** (a), **1_{Bu}** (b) and **2_{Bu}** (c). Experimental (black) and simulated (orange).

Table 1 Selected bond distances (average, Å) for **1_{Pr}**, **1_{Bu}**, and **2_{Bu}** with X-ray (esd in brackets) and computed DFT [Wiberg indices in brackets]

	1_{Pr} , X-ray	1_{Bu} , X-ray	1 , DFT	2_{Bu} , X-ray	2 , DFT
Pd–S	2.296(8)	2.297(3)	2.336 [0.351]	2.2600(11)	2.301 [0.260]
Ag _{ker} –S	2.7582(9)	2.718(3)	2.752 [0.132]	—	—
Ag _{cap} –S	2.9168(9)	2.8265(3)	3.040 [0.041]	—	—
Pd–Ag	2.8863(3)	2.8950(9)	2.916 [0.066]	2.9071(16)	2.997 [0.060]
Pd–Pd	—	—	—	2.9433(14)	2.985 [0.074]
Ag _{ker} –Ag _{ker}	3.0431(4)	3.0478(14)	3.124 [0.032]	—	—
Ag _{ker} –Ag _{cap}	3.1199(4)	3.1454(14)	3.215 [0.022]	—	—
Ag _{cap} –Ag _{cap}	3.1758(5)	3.2196(15)	3.482 [0.010]	3.039(18)	3.131 [0.040]

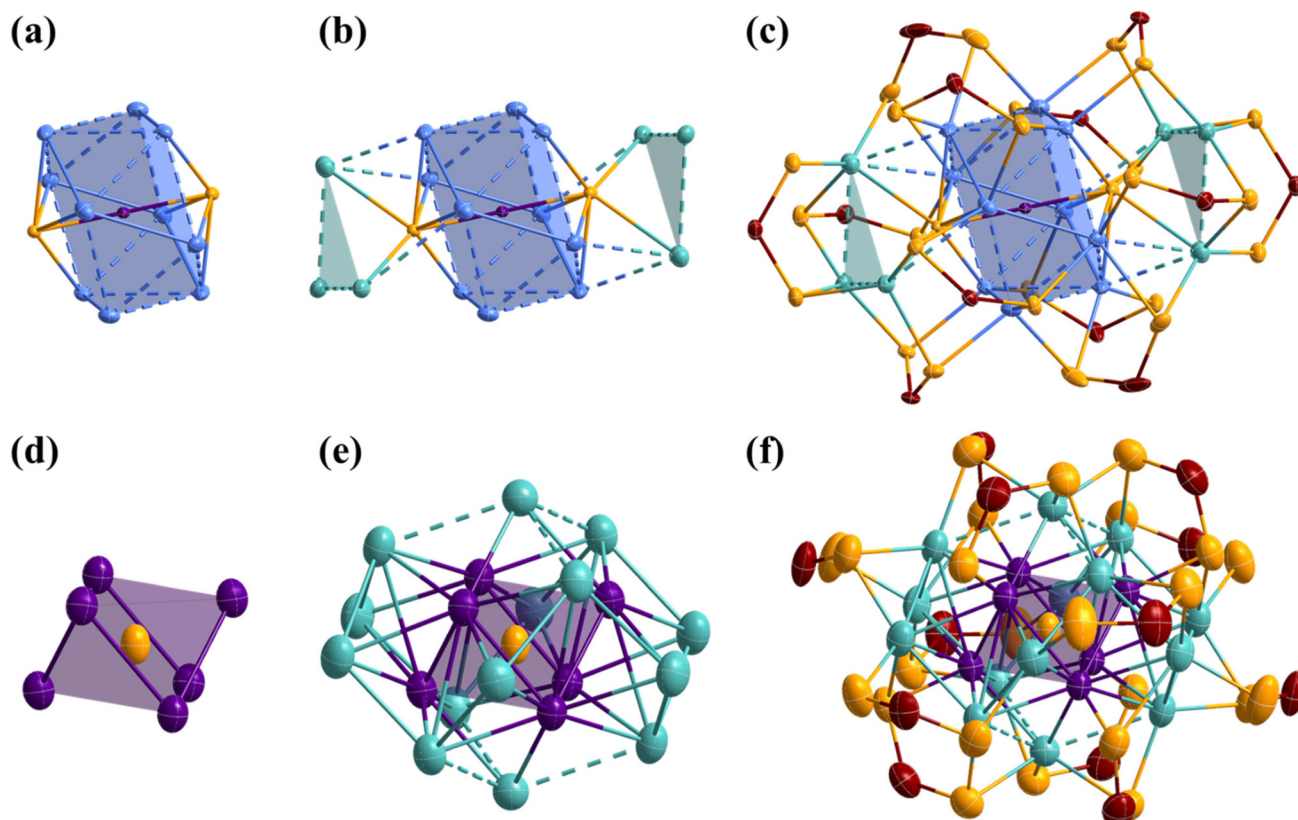


Fig. 2 The SCXRD structure of **1_{Bu}** and **2_{Bu}**. (a) The Pd@Ag₁₀S₂ kernel of **1_{Bu}**. (b) The two Ag₃ motifs capping the cluster core of **1_{Bu}**. (c) Total structure of **1_{Bu}** with ellipsoids drawn at 35%, H and C atoms omitted for clarity. (d) The S@Pd₆ core of **2_{Bu}**. (e) The S@Pd₆@Ag₁₄ framework of **2_{Bu}**. (f) Total structure of **2_{Bu}** with ellipsoids drawn at 35%, H and C atoms are omitted for clarity (color code: Pd, purple; Ag_{ker}, blue; Ag_{cap}, green; S, yellow; P, maroon).

atom. Another structural distinctive feature of **1** is the arrangement of the secondary silver layer around the Pd@Ag₁₀S₂ core. The six peripheral silver atoms are arranged into two triangular Ag₃ capping motifs, with only two atoms per triangle in bonding contacts with the Ag₁₀S₂ icosahedron (Fig. 2b). These triangular arrays are unprecedented in silver-rich NCs, where capping atoms usually occupy symmetrically favored positions dictated by the ligand shell. Consequently, the Ag_{cap}...Ag_{cap} distances in these peripheral motifs are notably longer than the Ag_{ker}...Ag_{ker} distances within the kernel. Two ligand coordination modes, six η^4 (μ_2, μ_2), and six η^3 (μ_2, μ_1), are observed in **1_{Pr}**. In contrast, **1_{Bu}** exhibits three coordination modes, namely two η^5 (μ_3, μ_2), four η^4 (μ_2, μ_2), and six η^3 (μ_2, μ_1). The increase in the denticity in **1_{Bu}** occurs for two ligands bridging the triangular Ag₃ motif to the kernel. This is accommodated by the adaptability of the sulfide, which is μ_7 in **1_{Bu}** compared to μ_8 in **1_{Pr}**. Notably, two of the six η^3 (μ_2, μ_1) in both **1_{Pr}** and **1_{Bu}** are located at unique positions on the triangular Ag₃ motifs of the nanocluster, where they do not interact with the central core (Fig. 2c).

The SCXRD structure of **2_{Bu}** is shown in Fig. 2 and selected metrical data are given in Table 1. **2_{Bu}** is found isostructural with the previously reported **2_{Pr}**.¹⁹ It shows a central sulfur

atom directing the NC shape through its encapsulation inside a Pd₆Ag₂ rhombohedron (distorted cube), the latter lying within an Ag₆ octahedron. This S@Pd₆Ag₂@Ag₆ framework is surrounded by six capping Ag atoms and twelve dtp ligands. The latter exhibit two coordination modes, namely six η^3 (μ_2, μ_1) and six η^4 (μ_2, μ_2) (Fig. 2f). The whole structure is of approximate *S*₆ symmetry. Each Pd atom in **2_{Bu}** is coordinated to the central sulfide ion and an S atom from a dtp ligand in a nearly linear fashion (S–Pd–S = 177°). In addition to two sulfur atoms, six Ag atoms and two Pd atoms complete the coordination sphere around each Pd atom. Notably, one of the Pd–Ag contacts is particularly long (~3.2 Å). The completely different metallic frameworks in **1** and **2** underscore the structural diversity achievable through ligand manipulation and the impact of varying metal ratios within these systems.

The ¹H NMR data confirm the presence of the isopropyl or isobutyl group in compounds **1** and **2**, as shown in Fig. S6c, d and S7b.† The **1_{Pr}** ³¹P NMR spectrum shows three distinct phosphorus environments identified by resonances at 105.31, 104.35, and 100.72 ppm. In contrast, **1_{Bu}** displays four resonances at 108.92, 106.27, 106.00, and 102.36 ppm (Fig. S6a and b†), indicating an additional coordination mode in **1_{Bu}**, in accordance with its solid-state structure. In the case of **2_{Bu}**, two

^{31}P NMR resonances are observed at 102.63 and 100.58 ppm, in agreement with the S_6 symmetry observed in the solid state being preserved in solution (Fig. S7a).†

Optical properties

NCs, due to their small size, typically fall below the size regime where surface plasmon resonance dominates their optical properties. However, subtle changes can significantly impact the discrete energy levels of these NCs. In general, as the size of the NC increases, the absorption spectra typically shift to longer wavelengths, attributed to the quantum confinement effect, wherein the energy levels become less discrete as the NC grows. However, larger NCs often exhibit lower photo-

luminescence quantum yields due to an increased opportunity for nonradiative decay processes. The optical spectra are shown in Fig. 3, and detailed values are compared in Table 2. Compounds **1_{Pr}** and **1_{Bu}** have similar absorption profiles with peaks observed at 365 and 366 nm (Fig. 3a). They emit in the NIR-I region, with peaks centered at 808 and 811 nm at 77 K, and emission lifetimes of 92 and 72 μs , respectively (Fig. 3b and S8a and b).† Compound **2_{Bu}** features a relatively broader absorption profile covering a range of 375–630 nm (Fig. 3a). Importantly, **2_{Bu}** exhibits unique NIR-II emission at ambient and low temperatures, with peaks centered at $\lambda_{\text{em}} = 1034$ nm (298 K) and $\lambda_{\text{em}} = 1004$ nm (77 K) (Fig. 3b). A change in the alkyl substituent in **2_{Pr}**, results in peaks centered at $\lambda_{\text{em}} =$



Fig. 3 (a) The absorption spectra for **1_{Pr}**, **1_{Bu}**, and **2_{Bu}** recorded in 2-MeTHF. (b) Emission spectra for **1_{Pr}** ($\lambda_{\text{em}} = 808$ nm), **1_{Bu}** ($\lambda_{\text{em}} = 811$ nm), **2_{Pr}** ($\lambda_{\text{em}} = 1007$ nm) and **2_{Bu}** ($\lambda_{\text{em}} = 1004$ nm) in 2-MeTHF at 77 K. (c) Comparison of λ_{em} for doped silver rich NC with entries labeled by the corresponding references.

Table 2 Spectroscopic information for series 1 and 2

	NC	λ_{max} (nm)	λ_{em} (nm)	τ (μs)	Φ_{em}^a (%)	k_o^b (μs^{-1})	k_r^c (μs^{-1})	k_{nr}^d (μs^{-1})
298 K	1_{Pr}	365	—	—	—	—	—	—
	1_{Bu}	366	—	—	—	—	—	—
	2_{Pr}	368, 408, 443, 548, 623	1046	0.0437	0.025	22.88	0.0057	22.87
	2_{Bu}	374, 413, 446, 545, 625	1034	0.127	0.294	7.87	0.023	7.85
77 K	1_{Pr}	—	808	92.3	—	—	—	—
	1_{Bu}	—	811	71.9	—	—	—	—
	2_{Pr}	—	1007	21.9	9.04	0.046	0.0041	0.042
	2_{Bu}	—	1004	20.6	14	0.049	0.0068	0.042

^a Φ_{em} = emission quantum yield (%). ^b k_o = mean excited-state decay rate constant = $1/\text{lifetime}$ (τ). ^c k_r = radiative constant = $\Phi_{\text{em}} \times 0.01 \times k_o$. ^d k_{nr} = nonradiative constant = $k_o - k_r$.

1046 nm (298 K) and $\lambda_{\text{em}} = 1007$ nm (77 K). Variations in the alkyl substituents of the ligand do not significantly influence the emission profile or lifetime. However, an increase in the size of the alkyl groups leads to an increase in the PLQY. This enhancement is possibly due to the increased electron-donating ability of the larger alkyl substituents, facilitating the emission process. The PLQYs for this series of NIR emissive compounds at ambient and low temperatures are listed in Table 2. The emission lifetimes of **2_{Pr}** and **2_{Bu}** increase at low temperatures, consistent with a spin-forbidden transition (Fig. S9).† A comparative analysis of emission maxima for doped Ag-rich nanoclusters reveals that NC **2** exhibits a significantly red-shifted emission wavelength compared to previously reported alloys^{25–30} and monometallic nanoclusters³¹ (Fig. 3c).

The cyclic voltammetry (CV) and the square wave voltammetry (SWV) of compounds **1** and **2** were recorded in organic solvent (THF and CH₂Cl₂) with 0.1 M [ⁿBuN][PF₆] as the supporting electrolyte. The results are summarized in Table 3 and Fig. 4 and S10–12.† At low temperature (233 K), the CV of **1_{Pr}** exhibits an irreversible reduction process (E_{pc}) at -1.23 V (vs. Fc⁺/Fc) and three irreversible oxidation processes (E_{pa}) at 0.17, 0.55 and 0.766 V, respectively (Fig. S10†). The SWV measurements confirm these redox processes, with $E_{\text{pc}} = -1.17, -1.51$, and -1.86 V and $E_{\text{pa}} = 0.25, 0.55, 0.94$, and 1.38 V at 298 K (Fig. 4a). **1_{Bu}** displays a similar pattern in both CV and SWV at

233 and 298 K in THF (Fig. S11†). The electrochemical HOMO–LUMO gap, which is calculated to be 1.40 eV, is different between O1 and R1. The results of **2_{Bu}** are shown at 298 and 233 K in Fig. 4b and S12.† Its calculated electrochemical HOMO–LUMO gap is 1.13 eV.

Bonding analysis and DFT investigation

The DFT-optimized geometry carried out on a simplified model for **1**, [PdAg₁₆S₂(S₂PH₂)₁₂], is in good agreement with its SCXRD structure (see Table 1 and corresponding xyz file in the ESI†). Analysis of its electronic structure supports the first-approximation view of a linear [PdS₂]^{4–} complex stabilized by a [Ag₁₆(S₂PH₂)₁₂]⁴⁺ cage, of which 12 Ag⁺ are in contact with the encapsulated [PdS₂]^{4–}. The NAO atomic charges (Table S2†) are consistent with Pd(0) and Ag(I) metal centers.¹⁹ Not considering the metal–metal contacts, 14 among the 16 Ag atoms are bonded to three dtp sulfur atoms, making them locally 16-electron centers. 4 of them not in contact with Pd (labelled Ag_{cap}) achieve a stable, nearly planar, coordination mode, whereas 10 of them in contact with Pd are somewhat pyramidalized to allow some 4d(Pd) → 5s/5p(Ag) electron transfer. Only 2 among the 16 Ag atoms are tetracoordinated, with a trigonal-planar coordination mode to three dtp sulfur atoms, to which a weaker bonding contact with one sulfide should be added. This bonding description is supported by the average Wiberg

Table 3 The redox potentials vs. Fc⁺/Fc(v) and calculated band gaps (eV)

NC	E_{pa}^a	E_{pc}^a	$E_{\text{g}}^{\text{E}b}$	$E_{\text{g}}^{\text{O}c}$	$E_{\text{g}}^{\text{PL}d}$
1_{Pr} (298 K)	+0.25, +0.55, +0.94, +1.38	−1.17, −1.51, −1.86	1.42	3.40	—
1_{Pr} (233 K)	+0.28, +0.61, +0.97	−1.35, −1.78	1.63	—	1.53
1_{Bu} (298 K)	+0.18, +0.41, +1.12	−1.22, −1.91	1.40	3.39	—
1_{Bu} (233 K)	+0.25, +0.95	−1.41, −1.89	1.66	—	1.52
2_{Bu} (298 K)	−0.23, +0.21, +0.78, +1.19	−1.36, −1.63, −2.20	1.13	1.98	1.20
2_{Bu} (233 K)	+0.01, +0.65, +1.09, +1.40	−1.52, −1.67, −2.04	1.49	—	1.24

^a Values determined from SWV spectra. ^b $E_{\text{g}}^{\text{E}} = E_{\text{pa}} - E_{\text{pc}}$. ^c $E_{\text{g}}^{\text{O}} = 1240/\lambda_{\text{max}}$, λ_{max} is the lowest electronic transition in absorption wavelength. ^d $E_{\text{g}}^{\text{PL}} = 1240/\lambda_{\text{em}}$, λ_{em} is the centered PL wavelength.



Fig. 4 Square wave voltammetry of **1_{Pr}** (a) and **2_{Bu}** (b) recorded in THF and CH₂Cl₂, at 298 K containing 0.1 M [ⁿBuN][PF₆] under a N₂ atmosphere.

bond indices (WBI) reported in Table S2,[†] notably the large Pd–S_{sulfide} value (0.351), the Ag_{ker}...Ag_{ker} and Ag_{ker}...Ag_{cap} values (0.032 and 0.022, respectively) characteristic of d¹⁰...d¹⁰ metallophilic interactions, and the somewhat larger Pd–Ag values (0.066) indicative of some weak covalent character in addition to metallophilicity. The orbital diagram of **1** is shown in Fig. S16.[†] It features two highest occupied orbitals derived from the two HOMOs of the hypothetical [PdS₂]^{4–}, with 4d_π(Pd) contribution mixed in an antibonding way with 3p_π(S). The lowest vacant orbitals are of 4s/4p(Ag) dominant character originating from the 16-electron Ag centers.

The TD-DFT-simulated UV-vis spectrum of **1** is shown in Fig. S18.[†] It shows a low-energy band at 502 nm, which is a combination of the HOMO → LUMO+1, HOMO–1 → L and HOMO–1 → LUMO+1 transitions, thus of 4d_π(Pd)/3p_π(S) → 5s/5p(Ag) nature. This low-energy peak is hardly distinguishable in the experimental spectra of **1_{Pr}** and **1_{Bu}** (Fig. 3a) and we believe that its computed oscillator strength is significantly overestimated in our TD-DFT calculations. The high-energy peak involves lower 4d(Ag) levels and the LUMO and LUMO+1, thus is of 4d(Ag) → 4s/4p(Ag) nature.

Although its composition and 3D architecture look fairly different from that of **1**, the bonding in **2**, whose simplified [Pd₆Ag₁₄(S)(S₂PH₂)₁₂] model we previously investigated,¹⁹ presents important similarities with that in **1**. Whereas **1** possesses a central linear S–Pd(0)–S motif, **2** shows six of them, sharing the central sulfide and making a flattened octahedron. Discarding the hypercoordinated nature of the sulfide, the six S–Pd(0)–S motifs are stable 14-electron centers, while the silver atoms are all tricoordinated 16-electron centers, of which those in contact with Pd are involved in (moderate) 4d(Pd) → 5s/5p(Ag) electron transfer. As in **1**, the whole architecture is maintained by the bidentate dtp ligands. The orbital diagram of **2** (Fig. S17[†]) shows highest occupied levels which can be roughly described as combinations of the two π-type HOMOs of the six S–Pd(0)–S motifs of 4d_π(Pd) and 3p_π(S) character, as in (**1**). Because of the existence of Pd...Pd contacts (Table 1), the 2 × 6 occupied combinations develop a “band” according to their (rather weak) amount of Pd...Pd bonding/antibonding character, of which the highest are (moderately) Pd...Pd antibonding, thus tending to limit the width of the HOMO–LUMO separation.

The TD-DFT-simulated UV-vis spectra of **1** and **2** (Fig. S18[†]) are in a good agreement with their experimental counterpart (Fig. 3a). They show two major bands which are of similar nature. Their major high-energy band is of 4d(Ag/Pd) → ligand character. The oscillator strength of lowest energy bands (of 4d_π(Pd)/3p_π(S) → 5s/5p(Ag) nature) are overestimated with respect to experiment. In fact, in the case of **1**, this computed low-energy band (502 nm, Fig. S18[†]) can be associated with transitions buried in the tails observed around 400–500 nm in the **1_{Pr}** and **1_{Bu}** experimental spectra (Fig. 3a). In the case of **2**, this band, also of 4d_π(Pd)/3p_π(S) → 5s/5p(Ag) nature, results from a transition from the degenerate HOMO–1/HOMO–2 e_g level to the degenerate lowest unoccupied e_g level (LUMO/LUMO+1). Although of similar nature, the low-energy band in **2** is red-shifted as compared to that computed for **1**. We suggest that this

is the same transition which is responsible for the NIR-II phosphorescence of **2**. The large Stokes shift experimentally observed can be rationalized from the fact that the e_u → e_g transition in **2** corresponds to the depopulation of a Pd...Pd antibonding level and the occupation of an Ag...Ag bonding level, with the consequence of a non-negligible stabilization of the corresponding excited state, a situation which does not exist in **1**.

Conclusion

This study has successfully demonstrated the strategic synthesis of sulfide-doped nanoclusters (NCs) by modulating the reaction pH with triethylamine, a method that effectively generates sulfide ions from dithiophosphate ligands. Notably, the isolation of **1_{Pr}** and **1_{Bu}**, which exhibit *the first silver-rich icosahedron containing two sulfur heteroatoms at its vertices*, resulted in the appearance of photoluminescence in the NIR-I region. Furthermore, the sulfide-centered NC **2_{Bu}** exhibits remarkable NIR-II photoluminescence at both ambient and low temperatures, highlighting the potential of this synthetic strategy for producing rare NIR emissive nanoclusters. Given that the NIR-II emission originates from a PdS₂ → Ag transition, the targeted incorporation of additional Pd and S atoms into silver NCs is a promising strategy for further enhancing their optical properties. This study sheds light on the intricate relationship between size, structure, and optical properties in NCs, emphasizing the importance of precise control over these parameters. This finding and our controlled synthesis and ligand modification capabilities open exciting avenues for designing new materials with tailored optical responses. These advancements hold promise for applications in bioimaging, sensing, optoelectronics, and beyond.

Author contributions

Y.-R. N., T.-H. C., and M. N. P.: investigation, data curation, formal analysis, methodology, and writing. J. Y. C., Y.-J. C., H. L., and S. K.: data curation and formal analysis. J.-Y. S.: data curation, formal analysis, and writing. C. W. L.: supervision, writing, conceptualization, project administration, and resources.

Data availability

Crystallographic data for [**1_{Pr}**, **1_{Bu}**, **2_{Bu}**] have been deposited at the Cambridge Crystallographic Data Centre under [CCDC 2379720–2379722].[†] Computational details (including coordinate file), X-ray structure analyses, NMR spectra, electrochemical data, ESI-MS data, and luminescence decay curves are available in the ESI.[†]

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work was supported by the National Science and Technology Council of Taiwan (113-2123-M-259-001) and the GENCI French National Computer Resource Center (A0030807367).

References

- 1 Y. Qin, X. Chen, Y. Gui, H. Wang, B. Z. Tang and D. Wang, *J. Am. Chem. Soc.*, 2022, **144**, 12825–12833.
- 2 A. Baghdasaryan, F. Wang, F. Ren, Z. Ma, J. Li, X. Zhou, L. Grigoryan, C. Xu and H. Dai, *Nat. Commun.*, 2022, **13**, 5613.
- 3 M. A. H. Muhammed, P. K. Verma, S. K. Pal, R. C. A. Kumar, S. Paul, R. V. Omkumar and P. Thalappil, *Chem. – Eur. J.*, 2009, **15**, 10110–10120.
- 4 Y. Xiao, Z. Wu, Q. Yao and J. Xie, *Aggregate*, 2021, **2**, 114–132.
- 5 Z. Luo, K. Zheng and J. Xie, *Chem. Commun.*, 2014, **50**, 5143–5155.
- 6 G. Yang, X. Pan, W. Feng, Q. Yao, F. Jiang, F. Du, X. Zhou, J. Xie and X. Yuan, *ACS Nano*, 2023, **17**, 15605–15614.
- 7 X. Song, W. Zhu, X. Ge, R. Li, S. Li, X. Chen, J. Song, J. Xie, X. Chen and H. Yang, *Angew. Chem., Int. Ed.*, 2021, **60**, 1306–1312.
- 8 Z. Liu, L. Luo and R. Jin, *Adv. Mater.*, 2024, **36**, 2309073.
- 9 Z. Liu, L. Luo, J. Kong, E. Kahng, M. Zhou and R. Jin, *Nanoscale*, 2024, **16**, 7419–7426.
- 10 H. Qian, M. Zhu, E. Lanni, Y. Zhu, M. E. Bier and R. Jin, *J. Phys. Chem. C*, 2009, **113**, 17599–17603.
- 11 A. Das, T. Li, K. Nobusada, Q. Zeng, N. L. Rosi and R. Jin, *J. Am. Chem. Soc.*, 2012, **134**, 20286–20289.
- 12 R. Jin, C. Liu, S. Zhao, A. Das, H. Xing, C. Gayathri, Y. Xing, N. L. Rosi, R. R. Gil and R. Jin, *ACS Nano*, 2015, **9**, 8530–8536.
- 13 X. Wan, W. W. Xu, S. Yuan, Y. Gao, X. Zeng and Q. Wang, *Angew. Chem.*, 2015, **127**, 9819–9822.
- 14 L. Luo, Z. Liu, X. Du and R. Jin, *J. Am. Chem. Soc.*, 2022, **144**, 19243–19247.
- 15 X. Fu, X. Lin, X. Ren, R. Wu, C. Liu and J. Huang, *Nanoscale*, 2020, **12**, 11825–11829.
- 16 M. S. Bootharaju, S. Lee, G. Deng, S. Malola, W. Baek, H. Häkkinen, N. Zheng and T. Hyeon, *Angew. Chem., Int. Ed.*, 2021, **60**, 9038–9044.
- 17 T. Jia, Z. J. Guan, C. Zhang, X. Z. Zhu, Y. X. Chen, Q. Zhang, Y. Yang and D. Sun, *J. Am. Chem. Soc.*, 2023, **145**, 10355–10363.
- 18 M. S. Bootharaju, S. M. Kozlov, Z. Cao, M. Harb, M. R. Parida, M. N. Hedhili, O. F. Mohammed, O. M. Bakr, L. Cavallo and J. M. Basset, *Nanoscale*, 2017, **9**, 9529–9536.
- 19 S. K. Barik, T.-H. Chiu, Y. C. Liu, M. H. Chiang, F. Gam, I. Chantrenne, S. Kahlal, J. Y. Saillard and C. W. Liu, *Nanoscale*, 2019, **11**, 14581–14586.
- 20 T.-H. Chiu, J.-H. Liao, F. Gam, I. Chantrenne, S. Kahlal, J. Y. Saillard and C. W. Liu, *J. Am. Chem. Soc.*, 2019, **141**, 12957–12961.
- 21 S. K. Barik, C. Y. Chen, T.-H. Chiu, Y.-R. Ni, F. Gam, I. Chantrenne, S. Kahlal, J. Y. Saillard and C. W. Liu, *Commun. Chem.*, 2022, **5**, 151.
- 22 Y.-R. Ni, M. N. Pillay, T.-H. Chiu, Y. Y. Wu, S. Kahlal, J. Y. Saillard and C. W. Liu, *Chem. – Eur. J.*, 2023, **9**, e202300730.
- 23 T.-H. Chiu, J.-H. Liao, Y. Y. Wu, J. Y. Chen, Y. J. Chen, X. Wang, S. Kahlal, J. Y. Saillard and C. W. Liu, *J. Am. Chem. Soc.*, 2023, **145**, 16739–16747.
- 24 Y.-R. Ni, M. N. Pillay, T.-H. Chiu, J. Rajaram, Y. Y. Wu, S. Kahlal, J. Y. Saillard and C. W. Liu, *Inorg. Chem.*, 2024, **63**, 2766–2775.
- 25 H. Yi, S. Song, S. M. Han, J. Lee, W. Kim, E. Sim and D. Lee, *Angew. Chem., Int. Ed.*, 2023, **62**, e202302591.
- 26 X. Liu, J. Yuan, C. Yao, J. Chen, L. Li, X. Bao, J. Yang and Z. Wu, *J. Phys. Chem. C*, 2017, **121**, 13848–13853.
- 27 T.-H. Chiu, J.-H. Liao, F. Gam, I. Chantrenne, S. Kahlal, J. Y. Saillard and C. W. Liu, *Nanoscale*, 2021, **13**, 12143–12148.
- 28 T.-H. Chiu, J.-H. Liao, F. Gam, Y. Y. Wu, X. Wang, S. Kahlal, J. Y. Saillard and C. W. Liu, *J. Am. Chem. Soc.*, 2022, **144**, 10599–10607.
- 29 Y.-R. Lin, P. V. V. N. Kishore, J.-H. Liao, S. Kahlal, Y.-C. Liu, M. H. Chiang, J. Y. Saillard and C. W. Liu, *Nanoscale*, 2018, **10**, 6855–6860.
- 30 W. J. Yen, J.-H. Liao, T.-H. Chiu, Y. S. Wen and C. W. Liu, *Inorg. Chem.*, 2022, **61**, 6695–6700.
- 31 R. S. Dhayal, Y. R. Lin, J.-H. Liao, Y. J. Chen, Y. C. Liu, M. H. Chiang, S. Kahlal, J. Y. Saillard and C. W. Liu, *Chem. – Eur. J.*, 2016, **29**, 9943–9947.
- 32 H. W. Chang, R. Y. Shiu, C. S. Fang, J.-H. Liao, P. V. V. N. Kishore, S. Kahlal, J. Y. Saillard and C. W. Liu, *J. Cluster Sci.*, 2017, **28**, 679–694.
- 33 N. Yoshinari, Z. L. Goo, K. Nomura and T. Konno, *Inorg. Chem.*, 2023, **62**, 9291–9294.
- 34 Y. M. Su, Z. Z. Cao, L. Feng, Q. W. Xue, C. H. Tung, Z. Y. Gao and D. Sun, *Small*, 2022, **18**, 2104524.
- 35 W. H. Wu, H. M. Zeng, Z. N. Yu, C. Wang, Z. G. Jiang and C. H. Zhan, *Chem. Commun.*, 2021, **57**, 13337–13340.
- 36 Z. A. Nan, Y. Xiao, X. Y. Liu, T. Wang, X. L. Cheng, Y. Yang, Z. Lei and Q. M. Wang, *Chem. Commun.*, 2019, **55**, 6771–6774.
- 37 S. Jin, S. Wang, Y. Song, M. Zhou, J. Zhong, J. Zhang, A. Xia, Y. Pei, M. Chen, P. Li and M. Zhu, *J. Am. Chem. Soc.*, 2014, **136**, 15559–15565.
- 38 W. D. Tian, W. D. Si, S. Havenridge, C. Zhang, Z. Wang, C. M. Aikens, C. H. Tung and D. Sun, *Sci. Bull.*, 2024, **69**, 40–48.
- 39 W. D. Si, C. Zhang, M. Zhou, Z. Wang, L. Feng, C. H. Tung and D. Sun, *Sci. Adv.*, 2024, **10**, eadm6928.
- 40 Z. Wang, Y. Wang, C. Zhang, Y. J. Zhu, K. P. Song, C. M. Aikens, C. H. Tung and D. Sun, *Natl. Sci. Rev.*, 2024, **11**, nwae192.