



Cite this: *Green Chem.*, 2024, **26**, 5744

Covalent porous catalysts for electrochemical reduction of CO₂

Shuanglong Lu,^a Hongyin Hu,^{†a} Huimin Sun,^{†a} Fulin Yang,^{†a} Han Zhu,^a Mingliang Du,^a Yinghua Jin^b and Wei Zhang^{*b}

The electrocatalytic CO₂ reduction reaction (eCO₂RR) stands out as a highly promising approach to simultaneously resolving the issue of the elevated atmospheric CO₂ concentration and its utilization to produce value-added products. One of the present challenges is the rational design of efficient catalysts toward the eCO₂RR and the understanding of their structure–activity relationship. Covalent porous catalysts (CPCs) with atomically dispersed active sites embedded in the porous organic backbone, including polymer-based catalysts, such as covalent organic frameworks (COFs) and conjugated microporous polymers (CMPs), and molecule-based catalysts, such as porous organic cages (POCs) have received increasing attention recently. They feature extremely high atom utilization, customizable porosity, a customizable backbone, and distinct electronic properties. In this review, the design principles of CPCs, both in their selection of metal sites (metallic centers, heterometallic centers and their axial coordination) and optimization of organic backbones (chelating sites, their electronic modulation and microstructures) will be covered, aiming to provide guidance to tune their eCO₂RR performance. Then representative examples will be discussed to reveal their structure–activity relationships. The perspective and key challenges in CPC based eCO₂RR electrocatalysts will be finally proposed.

Received 8th February 2024,
Accepted 8th April 2024

DOI: 10.1039/d4gc00717d

rsc.li/greenchem

1. Introduction

Due to the ongoing emissions from fossil fuels, carbon dioxide (CO₂) continues to build up in the atmosphere. According to the forecast by Mauna Loa Observatory in Hawaii, the global annual mean CO₂ concentration has exceeded 419 ppm in 2022, with an almost 100 ppm increase in the past 60 years.^{1–3} The rising trend of CO₂ concentration will be kept in the next few decades. The elevated atmospheric CO₂ concentration would give rise to a series of severe environmental crises, including ocean acidification, extreme weather events, the greenhouse effect, and species extinction.^{4,5} Meanwhile, CO₂ could be used as a useful feedstock to synthesize various valuable chemicals. Consequently, the capture and conversion of CO₂ into value-added products and fuels represents a key and meaningful route for the mitigation of these environmental problems and the utilization of carbon sources.^{6–11}

To date, some strategies have been proposed for the conversion of CO₂ into value-added products, including chemical

transformation,^{12–14} photocatalytic reduction,^{15–19} electrocatalytic reduction,^{20–23} and biological conversion.^{24–27} Among them, the electrocatalytic CO₂ reduction reaction (eCO₂RR) stands out as a highly promising pathway due to the following merits: (a) simple devices and mild reaction conditions; (b) high environmental compatibility; (c) possible combination of other renewable energy sources (*e.g.*, solar, wind, water, and biomass energy); and (d) highly tunable pathways to different products.^{18,28–30} However, how to effectively activate and selectively convert CO₂ is challenging, given its linear and chemically stable nature with poor electron affinity. The dissociation energy for breaking the C=O bond is higher than 750 kJ mol^{−1}. Therefore, a relatively large overpotential is required to initiate CO₂ activation, resulting in low energy efficiency. Furthermore, the eCO₂RR is a complex process involving multi-step complicated proton/electron transfer, yielding diverse gaseous and liquid products such as formic acid (HCOOH), carbon monoxide (CO), methanol (CH₃OH), methane (CH₄), ethylene (C₂H₄), and ethanol (CH₃CH₂OH).^{31–39} The thermodynamic redox potentials of these products are closely aligned, resulting in a limited selectivity for a specific product. Moreover, the eCO₂RR is invariably accompanied by the competing hydrogen evolution reaction (HER) due to their similar potential windows, which presents another significant obstacle restricting the catalytic efficiency and potential application.^{40,41} Thus, the rational

^aKey Laboratory of Synthetic and Biological Colloids, Ministry of Education, School of Chemical and Material Engineering, Jiangnan University, Wuxi 214122, Jiangsu, China

^bDepartment of Chemistry, University of Colorado Boulder, Boulder, CO 80309, USA. E-mail: wei.zhang@colorado.edu

[†]These authors contributed equally to this work.



design of electrocatalysts with high activity, selectivity, and stability becomes crucial in the practical eCO_2RR . Until now, many types of electrocatalysts have been developed for eCO_2RR , including noble metals, transition metals, metal oxides, metal alloys, and metal chalcogenides.^{42–51} Most of them could provide enhanced catalytic efficiency and considerable stability with increasingly clear structure–activity relationships.

To further enhance the utilization of active species and make it easier to illustrate the structure–activity relationships, atomically dispersed active sites are emerging as novel electrocatalysts toward the eCO_2RR .^{52–55} In particular, covalent porous catalysts (CPCs) with atomically dispersed active sites embedded in the porous organic skeleton have received increasing attention recently.^{56–60} They feature properties such as extremely high atom utilization, customizable porosity, a tunable backbone, and distinct electronic properties. Compared to classic electrocatalysts, many research studies have proved that porous catalysts could enrich CO_2 and enhance the atomic utilization of metals. Furthermore, the electronic structure of the active sites could be facily tailored. Different from the analogous Metal–Organic Frameworks (MOFs) with similar properties, CPCs are all formed *via* covalent bonds. They could endow them with exceptional structural stabilities.^{61–64} The CPCs here are mainly classified into two categories, one is polymer-based catalysts, such as covalent organic frameworks (COFs) and conjugated microporous polymers (CMPs), and the other is molecule-based catalysts, such as porous organic cages (POCs). They all have the versatility of their organic building blocks, organic linkers and variable side groups in their skeleton.^{59,65–70} Taking advantage of suitable chelating groups, diverse metallic sites could be incorporated into their architectures to serve as active sites toward the eCO_2RR .^{64,71–73} Moreover, their microstructures, such as their porosity and nanostructure, could also be effectively regulated in accordance with specific requirements.^{74–79} Apparently, these characteristics make CPCs an ideal platform for the development and fabrication of electrochemical CO_2RR catalysts.

Some excellent review articles on the porous frameworks for eCO_2RR have already been published.^{56,58,80–84} For example, Wang *et al.* recently published a comprehensive review on functional porous frameworks, including MOFs and COFs and their derived porous materials.⁵⁸ They summarized typical rational design strategies based on several key factors and different categories of materials. Nevertheless, other covalent porous catalysts (CPCs), such as conjugated microporous polymers (CMPs) and porous organic cages (POCs), were not covered. Moreover, with the rapid development in this area, more research studies have been published recently, which also need to be summarized timely. In this review, we mainly focus on several typical covalent porous catalysts toward the eCO_2RR , including COFs, CMPs and POCs. The design principles both for the selection of metal sites and optimization of organic backbones in CPCs will be covered, aiming to provide guidance to regulate their eCO_2RR performance (Fig. 1). Then



Fig. 1 Classification and design principle of the CPCs toward the eCO_2RR .

representative examples will be summarized to reveal their structure–activity relationships. Finally, the perspective and key challenges in CPC based eCO_2RR electrocatalysts will be proposed.

2. Design principles for CPCs with high eCO_2RR performance

CPCs were made by polymerization of organic monomers with typical chelating sites to anchor metal species as active sites. It is proposed that metal sites within CPCs are responsible for eCO_2RR activity, and through the tuning of organic building blocks, metal species and microstructures, their eCO_2RR activity could be efficiently regulated.^{79,85–87} In this section, we will first give a brief introduction on eCO_2RR , including the general mechanism, characteristic tools and fundamental parameters. Then we will classify the design principles into two categories, the metal catalytic sites and organic backbones, revealing the underlying factors that govern eCO_2RR performance in CPCs.

2.1 General introduction of the eCO_2RR

The eCO_2RR involves multi-step complicated proton/electron transfer reactions, which result in multiple products.^{31,88–91} The catalysis starts from the adsorption of CO_2 molecules, followed by the protonated intermediates, $^*\text{OCHO}$ or $^*\text{COOH}$. HCOOH could be obtained through another protonation step and desorption, while CO could form through further reduction, where $^*\text{CO}$ is a key intermediate during the eCO_2RR . Catalysts with a weak $^*\text{CO}$ binding energy usually produce CO as the main product. The strong binding energy of $^*\text{CO}$ may lead to the coverage of active sites (Fig. 2). Therefore,





Fig. 2 Possible pathways for CO₂ reduction over CPCs.

a suitable binding energy of *CO may induce subsequent protonation to give protonated *CHO and *COH, which will undergo further reduction to CH₃OH and CH₄. HCOH could be obtained through two step proton/electron transfer from the intermediate *OCHO or *COOH. It is worth noting that as only spatial separated metallic centers exist in CPCs, C₂+ products, such as CH₃COOH, CH₃CHO, CH₃CH₂OH, *etc.*, are rarely reported.

There are mainly three types of electrocatalytic reactors for the eCO₂RR, *i.e.*, an H-cell, flow cell and membrane electrode assembly (MEA) cell.^{92–94} The most commonly used reactor in fundamental studies is an H-cell. It is provided with low cost and easy operation. However, due to the limited solubility of CO₂ molecules in the aqueous electrolyte, the current density is usually low in an H-cell. The flow cell could achieve a relatively high current density by using the gas diffusion electrode, which also has broad applications in many laboratories. The MEA cell is quite close to the real energy conversion devices, which holds great potential for scalability and practical application, though it has not been widely applied in fundamental research. Electrochemical measurement methods, such as cyclic voltammetry (CV), linear sweep voltammetry (LSV) and chronoamperometry, are typically used to evaluate the activity and stability of electrocatalysts toward the eCO₂RR. Meanwhile, gaseous products, such as CO, H₂, CH₄, *etc.*, could be quantified by online or offline gas chromatography (GC), while liquid products, such as HCOOH, CH₃CH₂OH, *etc.*, could be quantified by nuclear magnetic resonance spectroscopy (NMR). Based on the analysis of the electrochemical curves and the quantitative results, several fundamental parameters are generally used for the comparison and evaluation of electrocatalysts' selectivity, activity and stability toward the eCO₂RR, including the onset potential, overpotential (η), Tafel slope, normalized current density, faradaic efficiency (FE), turnover number (TON), and turnover frequency (TOF).

2.2 Metal catalytic sites in CPCs

Isolated metal sites can be introduced into chelating sites before or after the synthesis of CPCs. Typically, the isolated

metal species are considered as the active sites in CPCs for eCO₂RR. Many previously reported works are devoted to understanding the origin of catalytic activity and designing excellent catalysts based on isolated metal centers. It should be pointed out that some metallic centers, such as Cu, could transform into metal clusters during electrocatalysis, which serves as actual active sites for the reaction. Even though it won't be discussed deeply here, more attention should be paid during the research to avoid misconceptions.

(1) Metallic centers. The selection of the metallic centers could provide facile regulation of CPCs toward the eCO₂RR.^{71,83} The metallic centers are believed to be one of the key factors that determine the catalytic activity and selectivity of CPCs. Typically, metallic centers are responsible for different affinities toward CO₂ molecules and diverse binding energies of adsorption intermediates during proton–electron transfer processes. The previously reported metallic centers in CPCs are mainly from the first transition series metals, such as Mn, Fe, Co, Ni, and Cu (Fig. 3a).

(2) Heterometallic centers. Although the mono-metallic centers in CPCs could already show relatively high catalytic performance toward the eCO₂RR, the kinetics of proton-coupled electron transfer and in-time release of the product still need to be enhanced.⁹⁵ The incorporation of bimetallic sites is believed to be plausible to improve this problem through the synergistic effects between bimetallic sites. Besides the heterometallic centers, the deliberately designed metal vacancies in the coordination sites are also proposed as efficient modulation strategies to influence the electrocatalytic performance of CPCs. The uncoordinated ligands could serve as both electron and proton regulators during the eCO₂RR.

(3) Axial coordination of metallic centers. The axial coordination of metallic centers represents one of the effective strategies to modulate the intrinsic adsorption behavior of intermediates during the eCO₂RR.^{96,97} It usually involves the introduction of M–X or M–O bonds in the axial direction of metallic active sites. This unique spatial coordination configuration endows metallic centers with a denser distribution of charges, enabling oriented electron transfer. The electronic configur-



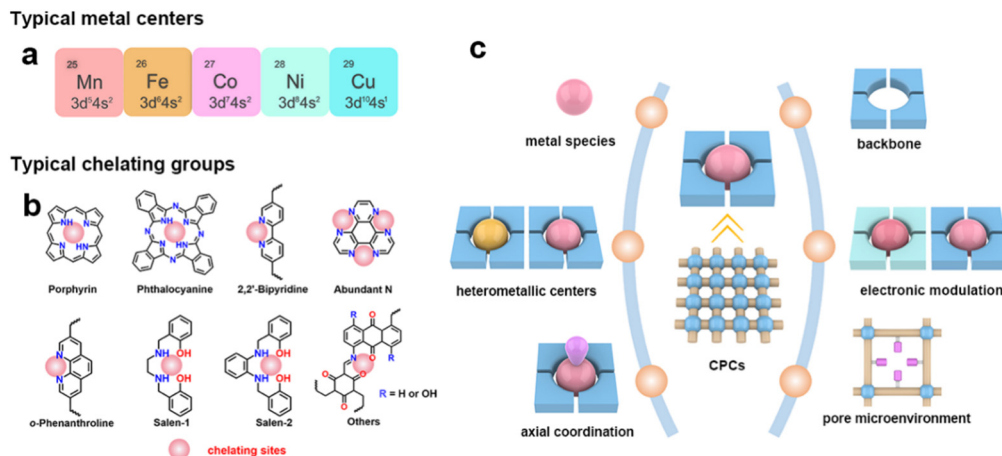


Fig. 3 Design principles of CPCs. (a) Typical metal centers and (b) typical chelating groups of CPCs toward the eCO_2RR ; (c) illustration of the design principle of the CPCs toward the eCO_2RR .

ation of metal centers is believed to exert a significant influence on the catalytic activity, selectivity, and stability of electrocatalysts.

2.3 Organic backbones in CPCs

To incorporate metallic centers inside CPCs, organic chelating sites with the binding ability to metal ions are needed.

(1) Chelating sites. Typical ligands such as porphyrin, phthalocyanine, salen, bipyridine, and 1,10-phenanthroline have been reported in previous research to coordinate with metal ions in CPCs (Fig. 3b). These chelating ligands are decisive to the local coordination structures of metallic centers, determining the preferred reaction pathways during the eCO_2RR .^{98–100} They usually have coordination atoms with lone pair electrons (e.g., N, O). The M-N_4 , especially in porphyrin/phthalocyanine-based CPCs, is the most commonly used symmetric chelating type with metal ions. Besides, the asymmetric $\text{M-N}_2\text{O}_2$ chelating type is typical in salen-based CPCs, and bipyridine-based or 1,10-phenanthroline-based CPCs with $\text{M}(\text{OAc})_x$ as the metal ion sources. When metal halides are used as metal precursors, the asymmetric $\text{M-N}_2\text{X}_2$ chelating type could also be constructed. The asymmetric coordination structures could break the uniform charge distribution on the metallic centers, inducing oriented electron transfer. These chelating sites with different metallic centers could provide an ideal platform for studying the structure–activity relationships during the eCO_2RR .

(2) Electronic modulation. As the organic backbones of CPCs are constructed with different organic building blocks, the electronic character of active sites could be tuned by the modification of reticular structures.^{101–103} Efficient charge transport along the skeleton's covalent backbone promotes the electronic connectivity between remote functional groups and the active sites, which could be used to modulate the catalytic performance of the metallic centers. The tunable sites could be located at the selected building blocks, suitable covalent linkages and variable side groups. More importantly, strategies such as establishing a comprehensive π -conjugated main

chain and constructing donor–acceptor heterojunctions are developed in previous research, as a result of which, the electron transfer capability could be effectively enhanced, together with the electronic modulation of the metallic centers.

(3) Microstructures. The microstructures of CPCs are typically determined by the inherent properties of the organic backbones, such as their porosity and nanostructure.^{104,105} Noteworthy, CPCs usually feature high porosity with adjustable pore sizes, pore distribution, and variable substituent groups in their cavities. These features have a great influence on the mass transfer and exposure of active sites during the electrocatalysis. According to the definition of the International Union of Pure and Applied Chemistry (IUPAC), the pore size could be classified into micropores (<2 nm), mesopores (2–50 nm) and macropores (>50 nm). It is proposed that a pore distribution between 1–10 nm is the highly suitable pore size range to facilitate the diffusion of CO_2 molecules and the penetration of the electrolyte. It is worth noting that even though compared to conventional catalysts, CPCs possess fully exposed active sites, not all electroactive sites in the porous structures could fully participate in electrocatalysis. The pore structure of CPCs is relatively intricate, leading to challenges during the eCO_2RR process, such as massive pH gradients, a variable local CO_2 concentration, and insufficient electron supply, which may hinder the complete utilization of active sites to a certain extent, particularly those located deeper inside the pores. To further enhance the catalytic efficiency, some nanostructures of CPCs, such as nanosheets, nanofibers or hollow tubes, could be obtained through morphology regulation. The elaborate design of these low dimensional nanostructures could ensure sufficient exposure and utilization of active sites.

3. Catalytic eCO_2RR by CPCs

3.1 COFs

COFs are one of the emerging and promising CPCs due to their high structure tunability, well-defined pore structures



and high specific surface areas. COFs can be designed and customized by selecting suitable building blocks at the molecular level toward the eCO_2RR . Introducing different chelating groups can provide various anchoring sites and coordination environments for different metal species.

In 2015, Chang and co-workers first reported the utilization of COFs as catalysts for eCO_2RR . A model framework (COF-366-Co) was constructed *via* an imine condensation between [5,10,15,20-tetrakis(4-aminophenyl)porphinato]cobalt [Co (TAP)] and 1,4-benzenedicarboxaldehyde (BDA) (Fig. 4).¹⁰⁶ The performance of the obtained catalysts was investigated in a buffer solution ($\text{pH} = 7.3$). Particularly, at a potential of -0.67 V, the catalyst demonstrated a FE_{CO} of 90%. The researchers proposed that a larger pore size would allow for higher electrochemical and chemical accessibility of the catalytic cobalt porphyrin active sites. They synthesized an extended COF-367-Co analogue by replacing BDA with biphenyl-4,4'-dicarbaldehyde (BPDA). It is then proved by CV studies that the extended organic framework COF-367-Co had a higher surface concentration of electroactive cobalt porphyrin sites compared to COF-366-Co. Moreover, given the low solubility of CO_2 , it is

speculated that not all cobalt porphyrin sites in the framework are fully engaged in electrocatalysis. Through the dilution of the active sites, the TOF values for CO generation could be significantly enhanced at the expense of decreased FE_{CO} .

This work has inspired the research interest in the field of COF-based electrocatalysts toward the eCO_2RR . Various research has been reported within the past few years. The metal species in COFs are the first critical factor, which deserve to be carefully considered. Their types and content could significantly determine the composition, faradaic efficiency and yield rate of products.

3.1.1 Metallic center design. As shown in Fig. 5a, Zhang *et al.* designed a series of nanosheet-like porphyrin-based COFs grown on amino-functionalized multi-walled carbon nanotubes (MWCNTs).¹⁰⁷ Different metal species (Co, Ni, Fe, Cu) were incorporated into the COFs through their chelation with the porphyrin. Notably, MWCNT-Por-COF-Co demonstrated remarkable activity ($\text{FE}_{\text{CO}} = 99.3\%$) and favorable durability in 0.5 M KHCO_3 toward the eCO_2RR among all the catalysts (Fig. 5b and c). While in 1.0 M KOH , MWCNT-Por-COF-Cu exhibited the highest FE_{CH_4} of 71.2% , which is also

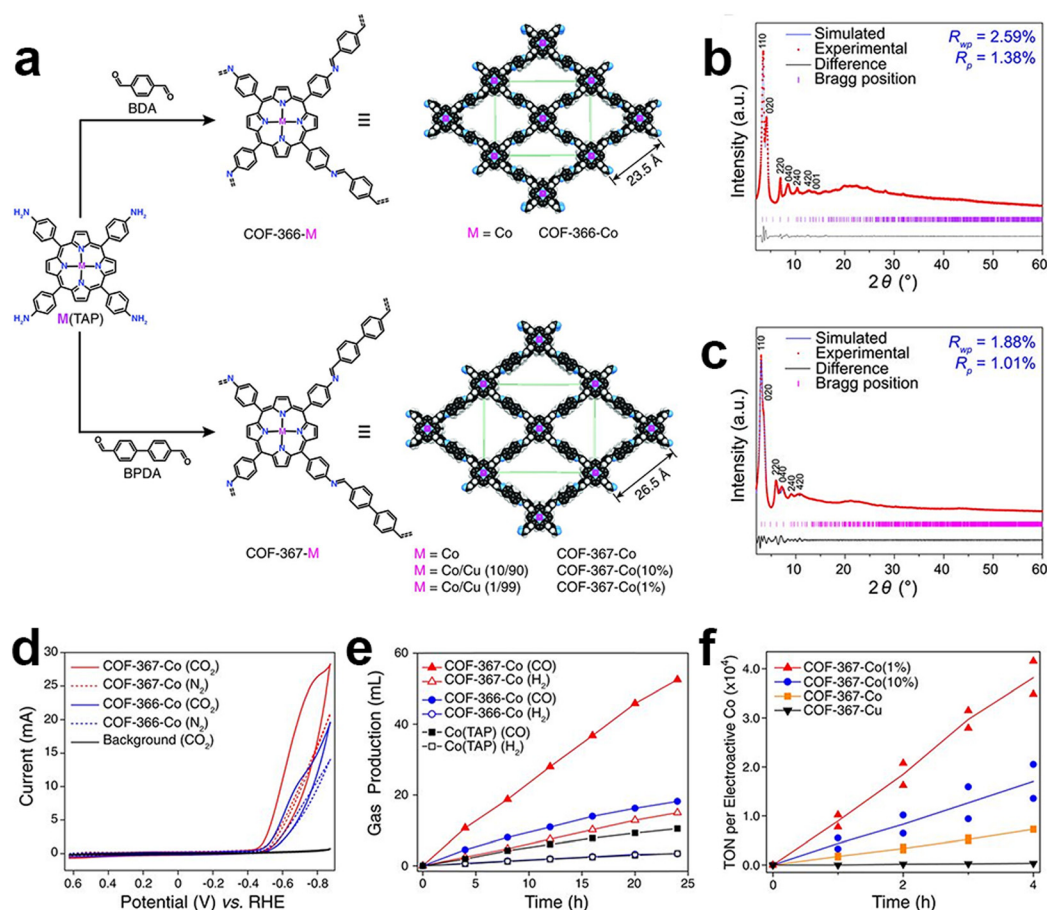


Fig. 4 Schematic illustration and performance of COF-366-M and COF-367-M. (a) Schematic illustration of the preparation of COF-366-M and COF-367-M; PXRD patterns of (b) COF-366-M and (c) COF-367-M; (d) cyclic voltammograms of COF-366-Co and COF-367-Co; (e) the volume of CO and H_2 produced by COF-367-Co, COF-366-Co and Co(TAP); (f) TON of CO production by COF-367-Co (1%), COF-367-Co (10%), COF-367-Co and COF-367-Cu.¹⁰⁶





Fig. 5 Schematic illustration of the metallic selection in COFs and their different performance. (a) Schematic illustration of the preparation of MWCNT-Por-COF and MWCNT-Por-COF-M; (b) LSV curves and (c) FE_{CO} of MWCNT-Por-COF-Fe, MWCNT-Por-COF-Co, MWCNT-Por-COF-Ni and Por-COF-Co; (d) FE of MWCNT-Por-COF-Cu; (e) the main product occupation of MWCNT-Por-COF-M.¹⁰⁷

the only catalyst which can convert CO₂ to CH₄ in this work (Fig. 5d). The detailed investigation indicated that the unique performance of MWCNT-Por-COF-Cu could be attributed to the copper-based nanoclusters generated during the electrocatalytic CO₂RR process.

In addition to the above-mentioned metallic sites, Mn ions could serve as the chelated metal centers toward the eCO₂RR. Julio Lloret-Fillol and colleagues incorporated a mononuclear Mn center into the pyridine-based COF (denoted as COF_{bpyMn}).¹⁰⁸ In this structure, the coordination geometry of Mn within the COF forms an octahedral structure, with three carbonyl groups facing the front and one bromine atom completing the first coordination shell. It is accepted that employing various coordination modes for metal centers can effectively modulate the electronic configuration of the metal centers. The embedded Mn species in the porous channels of COFs could change the redox pathway of the Mn molecular units, in terms of both steric hindrance, imposed by the rigidity of the COF structure, and the electronic effect of the organic framework. The redox process during the eCO₂RR was thus modulated compared with the molecular counterpart.

The COF_{bpyMn} material exhibited a low eCO₂RR onset potential (190 mV) and a high current density (>12 mA cm⁻², at an overpotential of 550 mV) in aqueous solution. Additionally, precise control of the electronic configuration of metal sites within COFs can be achieved through the axial coordination of metal centers. P. Kubiak *et al.* developed a simple bottom-up solvent-free synthesis method to prepare iron porphyrin COF films.¹⁰⁹ The iron centers in the porphyrin units were proposed to have an axially coordinated Cl. In this work, the molecular counterpart FeTAPPCl electrode demonstrated higher catalytic activity in the CV test, while the COF network could aid in the retention of FeTAPPCl molecules, leading to the increased CO production during electrolysis in organic electrolytes. Nevertheless, the influence of axial coordination on the electrocatalytic performance still needs to be further explored.

3.1.2 Organic backbone design: electronic modulation. Electron modulation has a multifaceted impact on catalytic performance, ranging from enhancing activity and fine-tuning selectivity. Besides the metallic species, the electronic modulation could be achieved through the meticulous backbone design and the incorporation of suitable functional groups.

Introducing alternate electron donor and acceptor building blocks into the skeleton of COFs can form a donor-acceptor network. It could endow the metallic centers with electronic-rich properties, which may benefit the activation of CO₂ molecules (Fig. 6a and b). Moreover, the electronic conductivity could also be enhanced in the donor-acceptor network. The Cao group¹¹⁰ and the Lan group¹¹¹ reported D-A COFs through the introduction of tetrathiafulvalene (TTF) into a deficit-electron cobalt porphyrin-based COF, respectively (Fig. 6c). It was proved that the D-A structures in the skeleton could amplify the inherent electron transfer capabilities of COFs. They also exhibited exceptional electrocatalytic performance toward the eCO₂RR with remarkable FE_{CO} and cycling stability. Another COFs with D-A structures were reported by the Cao group by replacing TTF with thieno[3,2-*b*]thiophene-

2,5-dicarbaldehyde (TT) with remarkable electron-donating properties (Fig. 6f).¹¹² Other reported alternatives include the viologen structure, all of which are conducive to the enhancement of their electronic conductivity and catalytic activity.¹¹³ These research studies provide a novel avenue for enhancing the current density of the eCO₂RR by integrating D-A hetero-junction moieties into the backbone of COFs, showcasing the positive influence of meticulous designed organic skeletons on the electrocatalytic performance. Liu *et al.* successfully synthesized a crown ether cobalt-porphyrin-based COF (TAPP(Co)-B18C6-COF).¹¹⁴ Within the constructed COF structure, the active metalloporphyrin core is tightly linked with 18-crown-6 units. This connection not only enhances the hydrophilicity of the framework scaffold but also facilitates the electron transfer process within the metalloporphyrin species. Furthermore, the

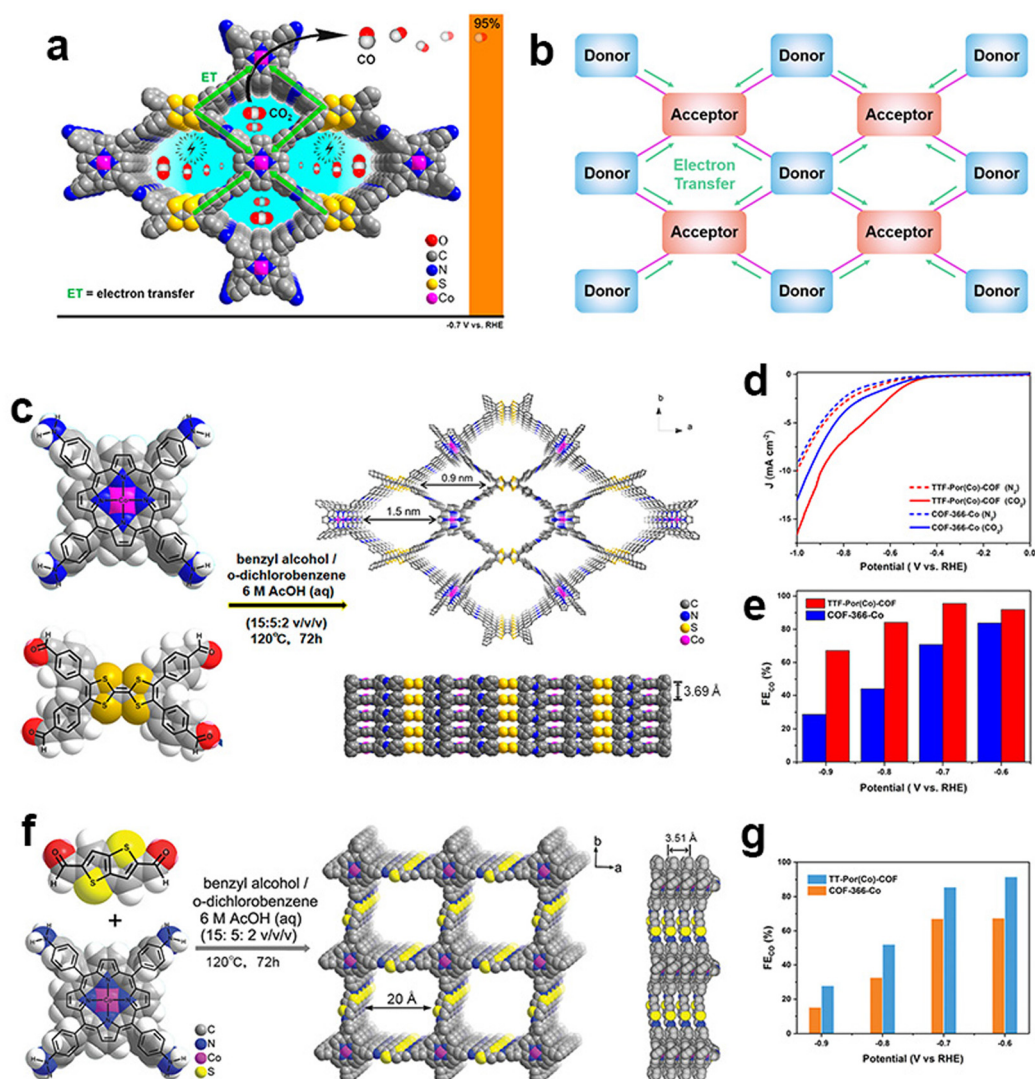


Fig. 6 Schematic illustration of the donor-acceptor network design of COFs. (a and b) Schematic illustration of electron transfer between donor blocks and acceptor blocks; (c) schematic illustration of the preparation of the TTF-Por(Co)-COF; (d) LSV curves of the TTF-Por(Co)-COF and COF-366-Co in the N₂-saturated and CO₂-saturated electrolytes; (e) FE_{CO} of the TTF-Por(Co)-COF and COF-366-Co;¹¹⁰ (f) schematic illustration of the preparation of the TT-Por(Co)-COF; (g) FE_{CO} of TT-Por(Co)-COF and COF-366-Co.¹¹²



presence of crown ether units bolstered the binding affinity for CO₂ molecules. This unique microenvironment provided by crown ether building blocks is highly favorable for the environment of the eCO₂RR performance. At −0.7 V, the TAPP(Co)-B18C6-COF demonstrated a FE_{CO} value of 93.3% and a maximum TOF of 1267 h^{−1}.

Other than the building blocks, the linkages in COFs could also exert an influence on the electronic states of metallic centers. The Zeng group employed a linking approach to synthesize two thiomorpholine-linked and dithiine-linked COFs with intriguing electronic and redox properties (Fig. 7a).¹¹⁵ Within these sulfur-containing heterocyclic materials, the low

electronegativity of sulfur atoms and their vacant d orbitals offer opportunities for electronic conjugation modulation (Fig. 7b–g). The dithiine-linked COF exhibited the highest TOF value of 1548.2 h^{−1} at −1.0 V toward CO production *via* the eCO₂RR. In comparison to the thiomorpholine-linked COF, the dithiine-linked COF not only displayed superior activity but also exhibited enhanced selectivity. It was proposed that a higher content of sulfur atoms contributed to higher conductivity and notable redox capability. Another work by the Zeng group reported a multi-tiered synthetic post-modification strategy to construct catalytic COFs for eCO₂RR (Fig. 7h).¹¹⁶ The imine linkage was converted into amine and ionic bonds was



Fig. 7 Schematic illustration of the linkage modulation of COFs. (a) The structure difference between thiomorpholine-linked and dithiine-linked COFs; (b) solid-state UV–vis absorption spectra and the (c) band structure diagram of thiomorpholine-linked and dithiine-linked COFs; (d) high-resolution XPS spectra of N 1s for the thiomorpholine-linked COF, dithiine-linked COF and CoPc. (e) XANES spectra of the Co K-edge, (f) the corresponding Co K-edge k_3 -weighted Fourier transform spectra and (g) WT contour spectra for the k_3 -weighted EXAFS data of the standard Co foil, CoPc, thiomorpholine-linked COF and dithiine-linked COF.¹¹⁵ (h) The synthesis of the N⁺-COF, NH-COF and N⁺-NH-COF from the base COF; (i) the UV-vis absorption (inset: Tauc plots) and (j) the energy gap (HOMO and LUMO) of the CoTAPP-PATA-COF (black), N⁺-COF (green), NH-COF (blue) and N⁺-NH-COF (red).¹¹⁶



introduced into COFs simultaneously ($N^+-NHCOF$ (N^+ : ionic modification; NH : reduction modification)). Through this dual modification process, the binding affinity of CO_2 and its electronic state are finely tuned, thereby enabling control over the activity and selectivity of the CO_2 reduction reaction. It is noteworthy that COFs possessing dual functional groups exhibit remarkable selectivity, achieving a maximum FE_{CO} of 97.32%, with a TOF value of $9922.68\ h^{-1}$.

3.1.3 Organic backbone design: electronic modulation by side chains. The substitution of side chains in the skeleton is also vital to the entire electron distribution in the COFs, thus modulating the electrocatalytic performance. Electron-withdrawing functional groups, such as halogen derivatives and nitro groups, guide the electron density away from the metal center, thus influencing its reactivity and selectivity. Conversely, electron-donating functional groups, such as amino and hydroxyl groups, provide additional electron density to the metal center, potentially enhancing its reduced

state and thereby modulating the catalytic activity of the reaction. Yaghi *et al.* synthesized a series of COFs with distinct electron-withdrawing groups, including COF-366-Co, COF-366-(OMe)₂-Co, COF-366-F-Co, and COF-366-(F)₄-Co (Fig. 8a).¹¹⁷ Experimental findings indicated that side groups with electronegative elements directly induced an electron-withdrawing effect on the cobalt center. The extent of this influence is proportional to the electronegativity and the quantity of the installed side groups. Observed electronic variations significantly altered the reactivity of molecular active sites within the material. Notably, in the context of judiciously optimizing CO_2 electroreduction catalysts, the potential of remote functionalization is advantageous as it mitigates the impact of introduced groups on electronic effects and potentially avoids interference from steric hindrance or non-covalent interactions. Similarly, the same strategy was utilized by Zhu *et al.*¹¹⁸ They polymerized different substituted benzaldehydes with cobalt porphyrin on amino-functionalized carbon nanotubes, resulting in a

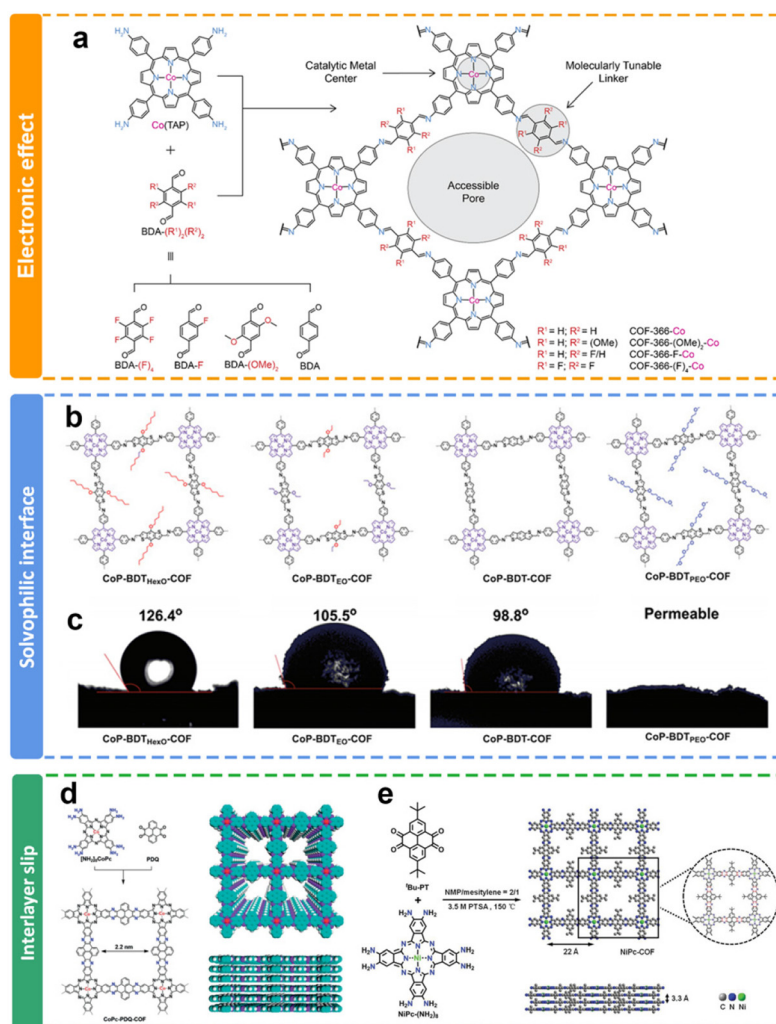


Fig. 8 Schematic illustration of the side chain modulation of COFs. (a) Design of cobalt-porphyrin derived covalent organic frameworks;¹¹⁷ (b) schematics and (c) images of contact angles of crosslinked metalloporphyrin networks with different electron-flow and mass-transport interfaces;¹¹⁹ (d) synthesis of phenazine-linked CoPc-based COFs;¹²⁰ (e) synthesis of phenazine-linked NiPc-based COFs with *tert*-butyl groups.¹²¹



series of COF-366-x-Co materials with varying substituents ($x = \text{H}; \text{OCH}_3; \text{F}; \text{OH}$). By introducing diverse functional groups, the electronic configuration of the metal sites within COFs, as well as the hydrophobicity and CO_2 affinity of the COFs, could be precisely tuned. Among these variants, COF-366-(OMe)₂-Co, containing methoxy functional groups, exhibited exceptional performance. At a potential of -1.05 V , it achieved a current density of 40 mA cm^{-2} . Within the voltage range of -0.58 to -0.88 V , the catalyst selectivity could be maintained at over 90%.

The internal environment of the COF channel, such as hydrophilic and hydrophobic, could be further improved through the selection of different side groups in the channels of COFs. Jiang and co-workers have adopted a bottom-up approach to integrate electronic conduction and mass transfer interfaces within the interconnected networks (Fig. 8b).¹¹⁹ Through topology-guided polycondensations under solvothermal conditions, crystalline solvophobic π networks with varying solvophobicities have been synthesized (Fig. 8c). The eCO_2RR involves the migration of CO_2 /solution through nano-channels to reach the localized reaction center within the local environment. It is noteworthy that both solvophobic and solvophilic interfaces were proved to enhance CO_2 absorption, while the solvophilic interface further increased water adsorption, leading to an increased occurrence of H_2 generation as a side reaction and subsequently reducing the faradaic efficiency of CO production.

Furthermore, the existence of side chain functional groups may cause large steric hindrance to induce interlayer slip, and prevent the overaccumulation of layers of COFs, which may benefit the exposure of active sites for eCO_2RR . The Jiang group¹²² and Cao group¹²¹ reported similar phenazine-linked Pc-based COFs (Fig. 8d and e). They all delivered remarkable performance toward the eCO_2RR due to their structural extended p-conjugation. Interestingly, the COFs in Cao's research has a *tert*-butyl side chain. The PXRD results demonstrated that the introduction of flexible side groups (*tert*-butyl groups) not only prevented the growth of large crystalline particles, but also gave rise to less-ordered edges. It is proposed that more active sites could be exposed due to the interlayer slip. It exhibits an impressively elevated CO selectivity exceeding 93% across a wide range of applied potentials spanning from -0.6 V to -1.1 V . Additionally, these COF nanosheets deliver a substantial j_{co} of 35 mA cm^{-2} at -1.1 V in aqueous solution.

3.1.4 Organic backbone design: porosity modulation. Pore size also plays a crucial role in the catalytic activity and selectivity of COFs due to the steric hindrance effect. Appropriate pore sizes can provide suitable reaction environments that facilitate interactions between catalytic active sites and reactants. In 2018, Jiang *et al.* investigated the influence of the pore size on the catalytic performance through theoretical calculations.¹²³ They revealed that at lower concentrations, CO_2 molecules tend to stay on the wall of COFs, particularly in the corner. With the increasing concentration of CO_2 , molecules will migrate toward central positions. It is claimed that smaller

pores of COFs may exhibit superior economic feasibility at atmospheric pressure.

Experimentally, Jiang *et al.* synthesized two polyimide-linked phthalocyanine COFs with different pore sizes through the polymerization between 2,3,9,10,16,17,23,24-octacarboxyphthalocyanine cobalt(II) tetraanhydrides with 1,4-phenylenediamine (CoPc-PI-COF-1) or 4,4'-diaminodiphenyl (CoPc-PI-COF-2) (Fig. 9b).¹²⁴ They have similar structures except for the pore size. Because of the identical Co(II) electroactive sites within the CoPc-PI-COFs, along with comparable permanent porosity and CO_2 adsorption capabilities, both frameworks exhibited analogous FE_{CO} values of 87% to 97% in a 0.5 M KHCO_3 solution when employed as cathodes at potentials ranging from -0.60 to -0.90 V . Nevertheless, CoPc-PI-COF-1 presented a j_{CO} of -21.2 mA cm^{-2} at -0.90 V and an elevated electrical conductivity relative to CoPc-PI-COF-2. The exclusive structural disparity between CoPc-PI-COF-1 and CoPc-PI-COF-2 lied in the longer bridging unit within CoPc-PI-COF-2, a factor potentially contributing to diminished conductivity in the latter and improved CO_2 concentration in the catalytic micro-environment. Similarly, another work by the Jiang group also demonstrated that the smaller pore size of COFs may be favorable for the enhancement of catalytic performance toward the eCO_2RR .¹²⁷ They successfully prepared two cobalt porphyrin-based COFs with pore sizes of 1.1 nm and 1.6 nm , respectively. They presented similar FE_{CO} values, while the COF with a smaller pore size exhibited more than 2-fold J_{CO} values when compared with the COF with a relatively larger pore size. They proposed that the larger charge-transfer resistance due to the longer electron transfer path caused by the additional benzene ring in the skeleton is possibly responsible for the relatively worse catalytic performance in the COF.

However, the opposite result has been reported by Huang's group (Fig. 9a). The pore size of CuPcF8-CoPc-COF (1.1 nm) is smaller than that of CuPcF8-CoNPc-COF (1.3 nm), while CuPcF8-CoPc-COF shows lower activity and selectivity ($\text{TOF} = 1.28 \text{ s}^{-1}$, $\text{FE}_{\text{COmax}} = 91\%$) than CuPcF8-CoNPc-COF ($\text{TOF} = 2.87 \text{ s}^{-1}$, $\text{FE}_{\text{COmax}} = 97\%$).¹²⁵ The authors attributed this to the larger pore size of CuPcF8-CoNPc-COF that affords higher CO_2 capacity within COF skeletons to enhance the accessibility of cobalt centers and the larger conjugation π - π stacking structure facilitates electron transfer during the catalytic process. In line with the result, Peng *et al.* also reported in their research that a COF with a larger pore size could deliver higher eCO_2RR performance (Fig. 9c).¹²⁶ They constructed COFs with different pore sizes through changing the nodes of building blocks. 2,3,9,10,16,17,23,24-Octacarboxyl phthalocyanine cobalt(II) (CoTAPc) as the node is coupled with 5,15-di(4-aminophenyl)-10,20-diphenylporphyrin (DAPor) or 5,15,10,20-tetra(4-aminophenyl)porphyrin (TAPor) to form two COFs with different pore sizes. The COF with a larger pore size demonstrated higher TOF values. This difference might be attributed to their distinct pore sizes, resulting in varying CO_2 mass transport capabilities.

The structural evolution in the dimension could significantly alter the microstructures of COFs. Compared to 2D





Fig. 9 Schematic illustration of the pore size modulation of COFs. (a and b) Schematic illustration of the modulation of the COF pore size *via* length building blocks;^{124,125} (c) schematic illustration of the modulation of the COF pore size *via* the nodes of building blocks.¹²⁶

COFs, 3D COFs possess a richer structural dimensionality, endowing them with greater porosity and surface area. This diversity in structure not only provides a broader space for the adsorption and diffusion of reactant molecules, but also creates numerous active sites. Recently, Cao *et al.* reported a 3D COF based on the condensation with tetra(4-formylphenyl) methane (TFPM), 5,10,15,20-tetrakis(4-aminophenyl)porphyrin cobalt (Co-TAPP), and 5,10,15,20-tetrakis(4-aminophenyl)porphyrin (TAPP) (Fig. 10a).¹²⁸ Within a voltage range of -0.6 to -1.1 V, CO and H₂ are the predominant products (Fig. 10b and c). Notably, the 3D COF exhibited a remarkable FE_{CO} of 92.4% at -0.6 V, surpassing the FE_{CO} of the 2D COF (82.4%). Furthermore, across a broader voltage window, the FE_{CO} could maintain over 80%. Another 3D COF was reported by Jiang *et al.*¹³⁰ They successfully synthesized a 3D polyimide-linked

phthalocyanine COF composed of interpenetrating pts nets (Fig. 10d). It was prepared *via* the polymerization reaction between 2,3,9,10,16,17,23,24-octacarboxyphthalocyanine tetra-anhydride M(TAPc) (M = Co, H₂) and 1,3,5,7-tetra(4-aminophenyl)adamantane (TAPA). Notably, the distinctive 3D porous structure had an impressive electrochemical surface concentration of 183 nmol cm⁻² on the electrode, accounting for approximately 32.7% of the total metal phthalocyanine sub-units. Consequently, a j_{CO} of 31.7 mA cm⁻² was achieved at 0.90 V, surpassing the performance of nearly all COF catalysts reported under similar conditions. Furthermore, this material demonstrated a TOF value of 0.6 s⁻¹ at 0.80 V during a continuous 20-hour test.

3.1.5 Organic backbone design: nanostructural modulation. Another important factor determining the microenvi-





Fig. 10 Schematic illustration of the synthesis of different 3D COFs and their performance. (a) Schematic illustration of prepare porphyrin-based 3D COF; (b) LSV curves and (c) FE of porphyrin-based 3D COF;¹²⁸ (d) Different kinds of phthalocyanine building blocks, (e) Schematic illustration of the preparation of phthalocyanine-based 3D COF.¹²⁹

ronment of metallic active species is the nanostructured morphologies of COFs, especially low dimensional nanostructures which could guarantee the sufficient exposure and utilization of active sites. Exfoliation is the mostly used strategy to obtain COF-based nanosheets. Cao *et al.* constructed metalloporphyrin-based COFs.¹¹³ The exfoliated COF nanosheets have a thickness of around 5.5 nm, which demonstrated more enhanced electrocatalytic performance (Fig. 11a–c). These superior attributes of the COFs can be attributed to the abundant accessible metallic sites and the efficient electron transfer processes. The strategy was also used by Lan *et al.* in the preparation of metalloporphyrin-tetrathiafulvalene based COF nanosheets (~5 nm) (Fig. 11d–f).¹¹¹ They found that upon exfoliation, the FE_{CO} can reach up to 99.7% at –0.8 V, higher than that of bulk COFs. A much thinner thickness of COFs (~0.7 nm) could be *in situ* achieved by the elaborate design of side chain functional groups in the skeleton, which have already discussed above (Fig. 11g and h).^{121,131} The interlayer slip may prevent the accumulation of layers of COFs due to steric hindrance. Another method to fabricate thin COF nanolayers rely on the assistance of additional catalyst supports, such as CNTs. As reported by Zhu *et al.* with the help of CNTs, the thickness of capped COF nanolayers was around 0.9 nm, equivalent to the thickness of three layers of 2D COFs (Fig. 11j).¹¹⁸ This ultra-thin COF nanolayer effectively shortened the electron transfer path from COFs to the conductive scaffold.

3.1.6 eCO₂RR into other products. The electrocatalytic reduction of CO₂ into various products necessitates distinct

proton-coupled electron transfer steps. As can be seen above, CO is the main product in most of the current research studies employing COFs as catalysts for the CO₂RR. The possible reason for that is the pathway from the reduction of CO₂ to CO only involves the transfer of two electrons and the removal of one O atom, while the further deep reduction of CO needs transfer of more electrons and proton coupling, which adds the difficulty. Moreover, in neutral and alkaline electrolytes, the absence of free protons poses challenges to efficient hydrogenation. Meanwhile, in acidic electrolytes, the hydrogen evolution reaction is highly competitive. Only a few research studies reported that other products could be generated with metallic COFs. The adsorption energy of intermediates on catalysts and the proton coupling process may be key factors in electrocatalyzing CO₂ reduction to other products besides CO. The electron transfer is accompanied by proton coupling and/or oxygen atom removal, and the sequence of proton and oxygen atom changes determines the type of product.

For instance, HCOOH and CO are both products from CO₂ reduction involving 2 electron transfer but through different pathways. Thus, designing COFs to effectively modulate the sequence of proton coupling and oxygen atom removal in the eCO₂RR process is beneficial for obtaining more valuable reduction products. Qiao *et al.* reported a Co-coordinated COF with a high nitrogen content (N-COF), which enables efficient CO₂ reduction to HCOOH.¹³² The COF was synthesized through direct condensation of hexaaminotriphenylene hexahydrochloride (HATP) and hexaketocyclohexane octahydrate



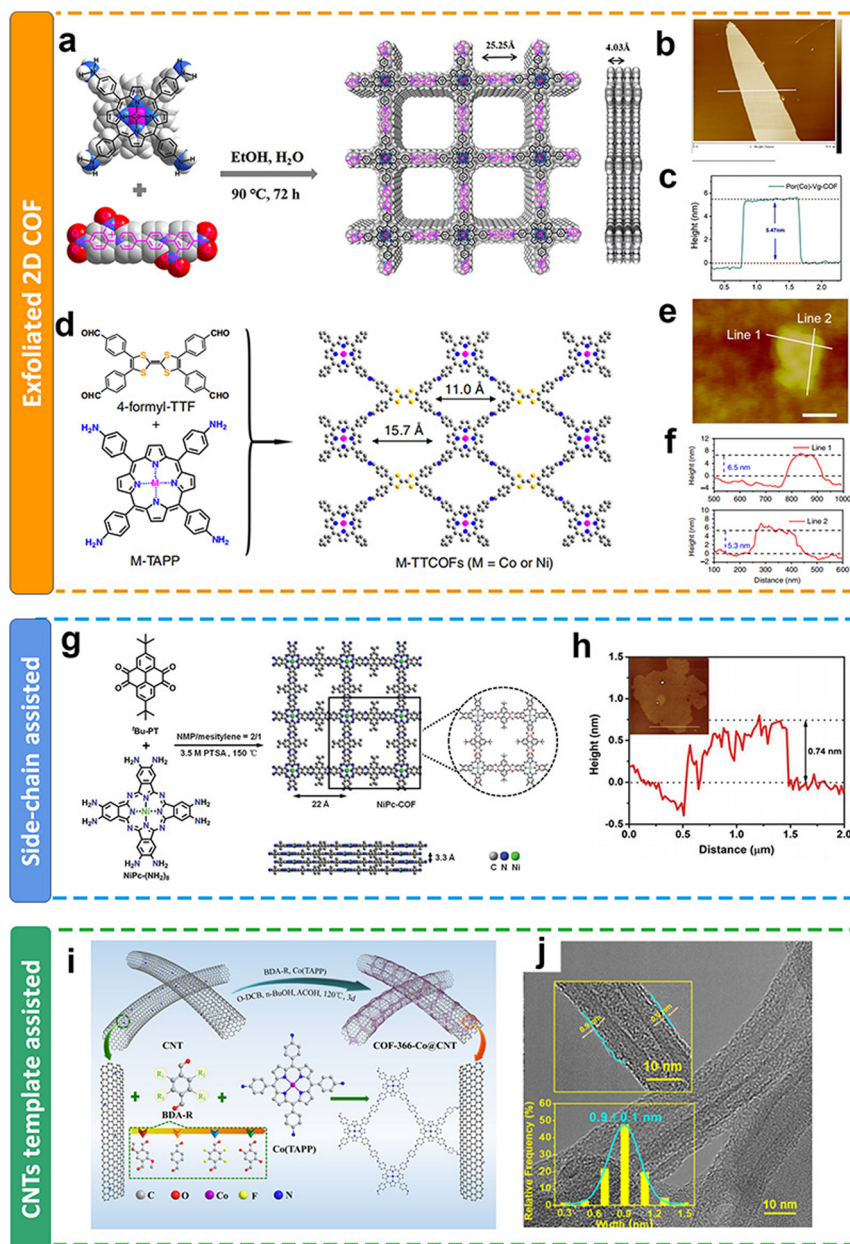


Fig. 11 Schematic illustration of the thickness modulation for 2D COFs. (a and d) Schematic illustration of the preparation of metalloporphyrin-based COFs; (b and e) AFM images and (c and f) the thickness of metalloporphyrin-based COFs after exfoliation;^{111,113} (g) schematic illustration of the preparation of metalloporphyrin-based COFs with side chain groups; (h) AFM images and the thickness of COFs with side chain groups after exfoliation;¹²¹ (i) schematic illustration of the preparation of metalloporphyrin-based COFs on CNTs; (j) TEM images and the thickness of metalloporphyrin-based COFs on CNTs.¹¹⁸

(HKH). The COF furnishes a coordinating environment for cobalt single atoms anchored between two nitrogen atoms. The coordination of cobalt in Co-N-COF reduces the bandgap significantly from 1.8 eV in N-COF to 0.67 eV, resulting in exceptional catalytic activity. The Co-N-COF exhibits remarkable activity with a J_{HCOOH} value of $\sim 446 \text{ mA cm}^{-2}$, a high FE_{HCOOH} value of 97.4%, and outstanding stability over 100 hours. Moreover, the purification of the electrocatalytically produced formic acid solutions reaches up to $\sim 100 \text{ wt\%}$. The

abundant N atoms around the Co species in the skeleton may be responsible for the production of HCOOH , a hydrogenated product also through two electron transfer.

It is interesting to find that CH_4 , as the complete reduction product involving proton coupling and oxygen atom removal, has been reported in several research studies. The metallic centers used in this research are all Cu species. It is proposed that the strong adsorption strength of typical intermediates on Cu species during the eCO_2RR may be the main reason for the



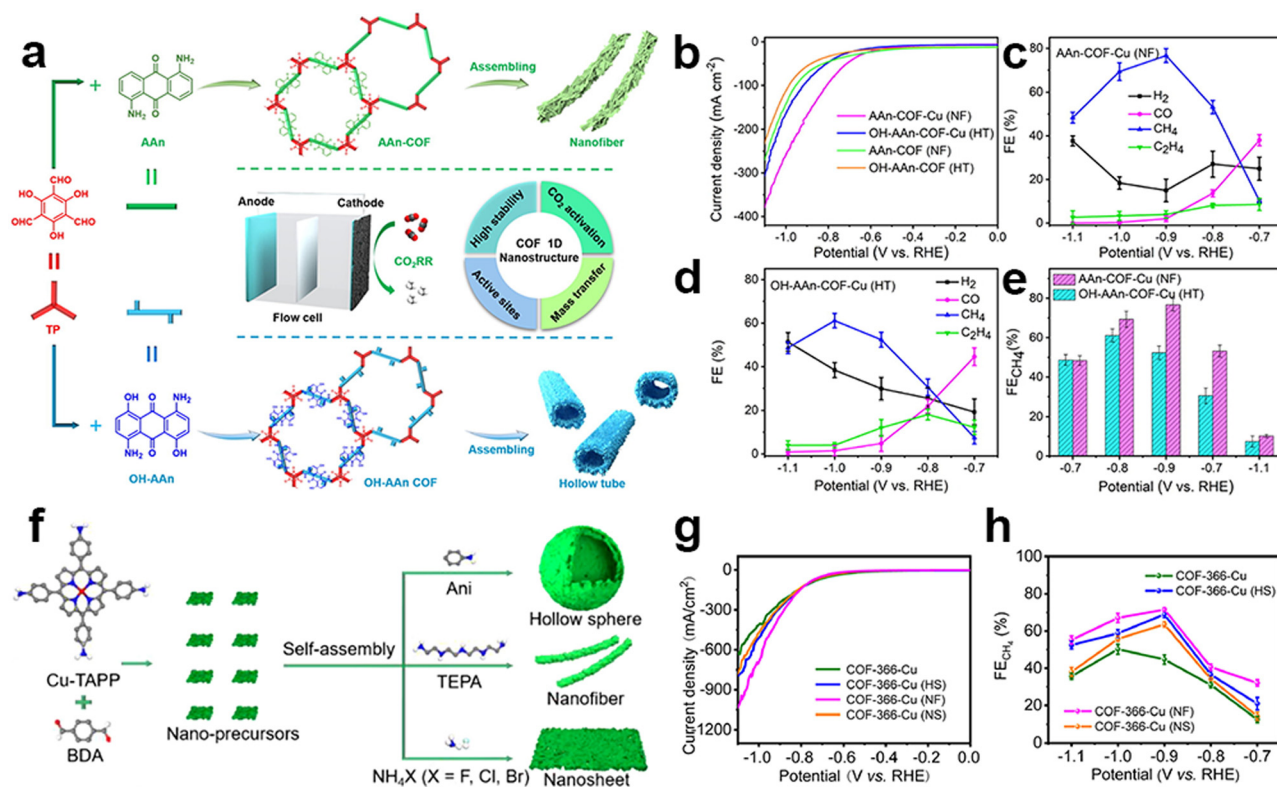


Fig. 12 Schematic illustration of super-structured COFs and their performance. (a) The schematic representation of anthraquinone COF-based nanofibers and hollow tubes in the electrocatalytic eCO_2RR ; (b) LSV curves of AAn-COF-Cu (NF), OH-AAn-COF-Cu (HT), AAn-COF (NF) and OH-AAn-COF (HT); (c–e) FE of AAn-COF-Cu (NF) and OH-AAn-COF-Cu (HT).¹³³ (f) The schematic representation of porphyrin COF-based nanostructures; (g) LSV curves and (h) FE_{CO} values of COF-366-Cu, COF-366-Cu (HS), COF-366-Cu (NF) and COF-366-Cu (NS).¹³⁴

production of this deeply reduced product. The Lan group reported anthraquinone-based COFs for efficient eCO_2RR . Tunable 1D superstructures (nanofibers and hollow tubes) of COFs could be achieved using different side groups in the building blocks, which were then post-modified with Cu ions within the N and O atoms (Fig. 12a).¹³³ These porous nanostructures are proposed to have the features of exceptional chemical stability, substantial surface area, and capability for CO_2 enrichment and activation. The Cu-incorporated COF nanofibers and hollow tubes exhibited superior faradaic efficiency for CH_4 production in flow cells, with a FE_{CH_4} value of 77% (at -0.9 V, 128.1 mA cm^{-2}) and 61% (at -1.0 V, 99.5 mA cm^{-2}), respectively (Fig. 12b–e). Notably, the study also found that the nanofiber morphology was more favorable for CH_4 production, while the hollow tube morphology had a particular control effect on the generation of ethylene, which provides a general methodology for exploring morphology-controlled COFs for electrocatalytic CO_2 reduction. The morphology-dependent catalytic performance was also reported by Lan *et al.* in the tailoring of porphyrin-based COFs incorporated with Cu ions (COF-366-Cu) (Fig. 12f).¹³⁴ They successfully prepared three distinct types of nanostructures, including hollow nanospheres (HS), nanosheets (NS) and nanofibers (NF). The nanostructured Cu-COF showed higher CH_4 selectivity (Fig. 12g and h). Specifically, COF-366-Cu (HS), COF-366-

Cu (NF) and COF-366-Cu (NS) exhibit a superior FE_{CH_4} of 68.2% (285.4 mA cm^{-2} at -0.9 V), 71.0% (350.8 mA cm^{-2} at -0.9 V) and 64.2% ($-276.0 \text{ mA cm}^{-2}$ at -0.9 V) in the flow-cell, respectively. Besides, the achieved $\text{FE}_{\text{CH}_4+\text{C}_2\text{H}_4}$ (82.8% at -0.9 V) of COF-366-Cu (HS) is superior to most of the reported COFs and copper-based electrocatalysts. Due to the unique catalytic mechanism of Cu, Cu single sites will convert into nano-clusters during the catalytic process which have been previously reported. The stronger adsorption energy between Cu clusters and CO intermediates can further reduce CO intermediates to more deeper reduction products on the active sites. This is why C_2H_4 could be produced in their works. Cu clusters, rather than chelated Cu metallic sites, are the real active sites during the catalysis, so this content won't be discussed further in this review.

Another work by the Lan group using exfoliation to synthesize large-scale (~ 1.0 mm) and ultrathin (~ 3.8 nm) nanosheets of Cu-based COFs toward the efficient eCO_2RR to CH_4 .¹³¹ The functionalized exfoliating agent, 2,4-diamino-6-chloro-1,3,5-triazine (Dct), could also serve as a co-catalyst during the eCO_2RR . It may enhance the CO_2 absorption/activation, stabilize active intermediates, and facilitate CO production, thereby increasing the concentration of CO around Cu active sites. The optimized COF structure comprises continuous units of Cu-porphyrin and Dct. At 0.90 V, the FE_{CH_4}



reaches 80%, with a current density of $\sim 220.0 \text{ mA cm}^{-2}$, which is nearly double that of the bare COF counterpart. The ultrathin layer of Cu-COFs was also achieved by the template-directed synthesis reported by the same group. They synthesized a series of honeycomb-like porous crystalline heteroelectrocatalysts by growing a copper-porphyrin-based COF on the surface of a porous MOF template. The specially designed heterostructures with an integrated porous MOF-template and ultrathin COF-coating enable efficient CO_2 adsorption/activation and conversion into CH_4 . The optimized heterogeneous structure delivered a superior FE_{CH_4} of 76.7%, with an excellent current density of 398.1 mA cm^{-2} .

In the synthesis of products with more than two C atoms, the C-C coupling step for ethylene and ethanol requires the coupling of $\ast\text{CO}$ with $\ast\text{CO}$ or $\ast\text{CHO}$, a process that necessitates adjacent active sites to facilitate its occurrence. Since most metal active sites in COFs are isolated, there have been few reports on the synthesis of products with more than two C atoms through the eCO_2RR using COFs. Recently, Chen *et al.* successfully synthesized acetic acid, a C2 product, using a PcCu-TFPN COF.¹³⁵ They proposed that during their synthesis, the C-C coupling step is achieved between $\ast\text{CH}_3$ and CO_2 . They compared PcCu-TFPN with copper single-atom catalysts (CuSAC) and copper-porphyrin COFs. According to the DFT calculations, the Cu atoms in PcCu-TFPN possess a higher electron density compared to CuSAC and Cu porphyrin, primarily due to the presence of more nitrogen atoms and stronger conjugation effects in the phthalocyanine structure. During the CO_2RR process, CuSAC exhibits unstable adsorption of $\ast\text{CO}$, with longer Cu-C bond lengths that promote the desorption of CO, making it difficult to proceed with further hydrogenation reactions. On the other hand, although the Cu porphyrin COF can also further hydrogenate $\ast\text{CO}$ to $\ast\text{CH}_3$, the higher electron density of Cu atoms in PcCu-TFPN , combined with the strong conjugation effect of phthalocyanine, leads to the transfer of d electrons from Cu to C, resulting in a lower oxidation state for C. Consequently, this strengthens the interaction between PcCu-TFPN and $\ast\text{CH}_3$. Such single-site catalytic centers could also help avoid the generation of unwanted byproducts like ethylene and ethanol, providing a feasible strategy for future eCO_2RR .

Typical COFs as electrocatalysts in the eCO_2RR are summarized in Table 1.

COF-based catalysts can obtain excellent eCO_2RR performance by selecting suitable building blocks due to the well-defined and tunable structures. However, COF-based catalysts also face several challenges. The first is that the stacking between adjacent layers of 2D COFs may block the metal catalytic sites, especially in porphyrin- and phthalocyanine-based 2D COFs. Exfoliating 2D COFs into oligolayer sheets or building 3D COFs are assumed to be conducive to make metal centers readily accessible. Moreover, the poor intrinsic electric conductivity limits the eCO_2RR activity of COFs. Developing highly conductive COFs or incorporating conductive carbon materials could help improve the intrinsic electric conductivity of COFs.

3.2 CMPs

CMPs represent an important class of amorphous organic porous materials. Since their discovery in 2007, CMPs have become an important subclass of porous materials, which combine the characteristics of extended π -conjugated polymers and permanent microporous scaffolds. They have the advantages of large specific surface area, high stability, and tunable nanostructure. Well-designed CMPs can selectively adsorb CO_2 , and much research has been devoted to exploring their applications in electrocatalytic CO_2 reduction. Moreover, compared with COFs, their extended π -conjugated structures could deliver relatively better electrical conductivity, which is highly required in electrocatalysis.

CMPs could provide rigid frameworks to accommodate electroactive metal species to effectively catalyze the eCO_2RR . For example, Alexander J. Cowan *et al.* synthesized a bipyridine-based amorphous polymer CMP through the Sonogashira-Hagihara cross-coupling reaction of 1,3,5-triethynylbenzene with 5,5'-dibromo-2,2'-bipyridine and 1,4-dibromobenzene.¹⁴⁴ Then the metal-organic CMP containing a manganese carbonyl electrocatalyst was obtained after the reflux with $[\text{Mn}(\text{CO})_5\text{Br}]$. $[\text{Mn}(\text{bpy})(\text{CO})_3\text{Br}]$ and its derivatives have previously been reported to be active toward the eCO_2RR , while it will undergo dimerization prior to formation of the catalytically active species in solution. It was proved that after fixing it in the CMP framework, the dimerization could be prevented. According to CV and LSV curves, the Mn center of the complex remained electrochemically active within the CMP framework with CO as the main product.

In addition to the rigid framework of CMPs as a support, it is also widely believed that the highly conjugated structure in CMPs is conducive to electron transfer. Duan *et al.* prepared a two-dimensional conjugated nickel phthalocyanine polymer (NiPcP) by a simple one-step vacuum calcination method.¹⁴⁵ In comparison with the NiPc monomer, NiPcP possesses a highly cross-linked conjugated network structure, a higher surface area, more accessible active sites and enhanced structural robustness. Consequently, NiPcP exhibits a remarkably high performance toward the eCO_2RR to CO with a high $\text{FE} > 98\%$ in a wide potential range from -0.15 to -0.60 V . The maximum partial current density (j_{CO}) reaches 236 mA cm^{-2} at -0.6 V , far superior to the NiPc monomer. Experimental and theoretical studies revealed that the cross-linked conjugated structure was responsible for its high performance in the CO_2RR . Utilizing carbons such as graphene and carbon nanotubes (CNTs) to support CMPs is an effective way to improve electrical conductivity. However, the catalytic activity of MPc-containing CMPs often suffers from a limited molecular modulation strategy. As for NiPc, there is a strong correlation between the localized d electrons of the Ni atom and delocalized π electrons of the phthalocyanine ligand. The expansion of π -conjugation is expected to regulate the electron density of the Ni position. Liu *et al.* established nickel polyphthalocyanine (NiPPc) supported on carbon nanotubes (NiPPc/CNTs) for efficient eCO_2RR (Fig. 13a).¹⁴⁶ The XRD pat-



Table 1 Typical COFs as electrocatalysts in the eCO₂RR

Main product	Material	Metal center	Chelating site	Design principle	Highest FE	TOF	Reaction conditions	Ref.
CO	COF-366-Co COF-367-Co	Co Co and Cu	Porphyrin	Microstructures and heterometallic centers	90% 70%	~2500 h ⁻¹ ~4400 h ⁻¹	H type cell and 0.5 M KHCO ₃	106
CO	FeDhaTph-COF	Fe	Porphyrin	Electronic modulation	80%	>600 h ⁻¹	Five-necked pear-shaped flask, 0.1 M TBAPF ₆ and 0.5 M TFE MeCN solution	109
CO	TTF-Por(Co)-COF	Co	Porphyrin	Electronic modulation	95%	676 h ⁻¹	H type cell and 0.5 M KHCO ₃	110
CO	CoPc-PDQ-COF	Co	Phthalocyanine	Electronic modulation	96%	11 412 h ⁻¹	H type cell and 0.5 M KHCO ₃	122
CO	COF-366-Co@CNT COF-366-(OMe) ₂ -Co@CNT COF-366-(OH) ₂ -Co@CNT COF-366-(F) ₄ -Co@CNT	Co	Porphyrin	Electronic modulation and microstructures	92% 94% 90% 90.50%	>10 000 h ⁻¹ 11 877 h ⁻¹ ~10 000 h ⁻¹ ~8000 h ⁻¹	H type cell and 0.5 M KHCO ₃	118
CO	COF-366-Co COF-366-(OMe) ₂ -Co COF-366-F-Co COF-366-(F) ₄ -Co	Co	Porphyrin	Electronic modulation and microstructures	87% N/A N/A N/A	N/A N/A N/A N/A	H type cell and 0.5 M KHCO ₃	117
CO	2D NiPc-COF	Ni	Phthalocyanine	Electronic modulation	99.10%	3772 h ⁻¹	H type cell and 0.5 M KHCO ₃	121
CO	2D TT-Por(Co)-COF	Co	Porphyrin	Electronic modulation	91.40%	481 h ⁻¹	H type cell and 0.5 M KHCO ₃	112
CO	COF _{bpyMn}	Mn	2,2-Dipyridine	Electronic modulation and microstructures	62%	627 h ⁻¹	H type cell and 0.5 M NaHCO ₃	108
CO	CoPc-PI-COF-3	Co	Phthalocyanine	Electronic modulation and microstructures	96%	0.6 s ⁻¹	H type cell and 0.5 M KHCO ₃	130
CO	CuPcF ₈ -CoPc-COF CuPcF ₈ -CoNPc-COF	Cu and Co	Phthalocyanine	Heterometallic centers and microstructures	91% 97%	1.25 s ⁻¹ 2.87 s ⁻¹	H type cell and 0.5 M CsHCO ₃	125
CO	3D-Por(Co/H)-COF	Co	Porphyrin	Microstructures	92.40%	4610 h ⁻¹	H type cell and 0.5 M KHCO ₃	128
CO	CoPc-2H ₂ Por COF CoPc-H ₂ Por COF	Co	Phthalocyanine	Heterometallic centers, microstructures, and electronic modulation	95% 94%	>800 h ⁻¹ >400 h ⁻¹	H type cell and 0.5 M KHCO ₃	126
CO	CoP-BDT _{HexO} -COF CoP-BDT _{EO} -COF CoP-BDT-COF CoP-BDT _{PEO} -COF	Co	Porphyrin	Microstructures and electronic modulation	>90% >90% >90% ~60%	8655 h ⁻¹ 5425 h ⁻¹ 3336 h ⁻¹ 1351 h ⁻¹	H type cell and 0.5 M KHCO ₃	119
CO	TPPDA-CoPor-COF TPPDA-NiPor-COF	Co Ni	Porphyrin	Metal species and electronic modulation	92% 76%	2.0 s ⁻¹ ~0.4 s ⁻¹	H type cell and 0.5 M KHCO ₃	136
CO	Por(Co)-Vg-COF	Co	Porphyrin	Electronic modulation	91%	N/A	H type cell and 0.5 M KHCO ₃	113
CO	TPE-CoPor-COF TPTPE-CoPor-COF	Co	Porphyrin	Electronic modulation and microstructures	95% 96%	2.4 s ⁻¹ 1.21 s ⁻¹	H type cell and 0.5 M KHCO ₃	127
CO	TAPP(Co)-B ₁₈ C ₆ -COF	Co	Porphyrin	Electronic modulation and microstructures	93.20%	1267 h ⁻¹	H type cell and 0.5 M KHCO ₃	114
CO	NiPc-TFPN COF CoPc-TFPN COF ZnPc-TFPN COF	Ni Co Zn	Phthalocyanine	Metal species and electronic modulation	99.8% 96.1% 22.9%	490 h ⁻¹ 369 h ⁻¹ ~50 h ⁻¹	H type cell and 0.5 M KHCO ₃	137
CO	CoPc-DNDS-COF CoPc-DSDS-COF	Co	Phthalocyanine	Electronic modulation	96.30% 96.50%	1076.4 h ⁻¹ 1548.2 h ⁻¹	H type cell and 0.5 M KHCO ₃	115
CO	CoPc-PI-COF-1 CoPc-PI-COF-2	Co	Phthalocyanine	Electronic modulation and microstructures	97% 96%	2.2 s ⁻¹ 1.9 s ⁻¹	H type cell and 0.5 M KHCO ₃	124
CO	CoTAPP-PATA-COF N ⁺ -COF NH-COF N ⁺ -NH-COF	Co	Porphyrin	Electronic modulation	81.75% 92.07% 95.26% 97.32%	4166.19 h ⁻¹ 7453.19 h ⁻¹ 7604.63 h ⁻¹ 9922.68 h ⁻¹	H type cell and 0.5 M KHCO ₃	116
CO	Co-TTCOFs Ni-TTCOFs	Co Ni	Porphyrin	Electronic modulation and metal species	91.3% 20.9%	1.28 s ⁻¹ N/A	H type cell and 0.5 M KHCO ₃	111
CO	H ₂ Pc-COFs 0.25 NiPc-COFs 0.5 NiPc-COFs 0.75 NiPc-COFs NiPc-COFs	Metal free Ni	Phthalocyanine	Electronic modulation and heterometallic centers	34.21% 37.66% 92.52% 95.37% 75.68%	1258.64 h ⁻¹ 2790.41 h ⁻¹ 4713.53 h ⁻¹ 3749.88 h ⁻¹ 3860.12 h ⁻¹	H type cell and 0.5 M KHCO ₃	138
CO	Co-pPPC@CNFs Co-anPPC@CNFs	Co	Phthalocyanine	Electronic modulation and microstructures	74% 67%	N/A N/A	H type cell and 0.5 M KHCO ₃	139
CO	PcNi-im PcNi-pz PcNi-tfpn	Ni	Phthalocyanine	Electronic modulation	100% 98% 99%	3.06 s ⁻¹ 2.37 s ⁻¹ 2.45 s ⁻¹	Flow cell 0.5 M KHCO ₃	140
CO	Co-iBFBim-COF-I ⁻	Co	Porphyrin	Electronic modulation	99%	3018 h ⁻¹	MEA cell and 1 M KOH	141
CO	Co-Bpy-COF-Ru _{1/2}	Co	Porphyrin and 2,2-dipyridine	Electronic modulation	84.6%	681 h ⁻¹	H type cell and 0.5 M KHCO ₃	142



Table 1 (Contd.)

Main product	Material	Metal center	Chelating site	Design principle	Highest FE	TOF	Reaction conditions	Ref.
CO	pCoPc	Co	Phthalocyanine	Electronic modulation and metal species	67%	N/A	Flow cell and 1 M KOH	143
	pNiPc	Ni			99%	N/A		
	pCoNiPc	Co and Ni			>90%	N/A		
	pCo ₁ Ni ₆ Pc	Co and Ni			>90%	N/A		
CO	MWCNT-Por-COF-Fe	Fe	Porphyrin	Metal species	~63%	>25 s ⁻¹	H type cell and 0.5 M KHCO ₃	107
	MWCNT-Por-COF-Co	Co			99.30%	70.6 s ⁻¹		
	MWCNT-Por-COF-Ni	Ni			~60%	>20 s ⁻¹		
CH ₄	MWCNT-Por-COF-Cu	Cu			71.20%	N/A	H type cell and 0.1 M KOH	
CH ₄	Cu-Tph-COF-Det	Cu	Porphyrin	Microstructures and electronic modulation	~80%	N/A	Flow cell and 0.1 M KOH	131
CH ₄	Cu-Tph-COF-OH	Cu	Porphyrin		48.8%	N/A	Flow cell and 0.1 M KOH	
CH ₄	AAn-COF-Cu	Cu	Others		77%	N/A	Flow cell and 1 M KOH	133
CH ₄	OH-AAn-COF-Cu	Cu	Others		61%	N/A	Flow cell and 1 M KOH	
CH ₄	COF-366-Cu-HS	Cu	Porphyrin	Microstructures	68.20%	N/A	Flow cell and 1 M KOH	134
	COF-366-Cu-NF	Cu	Porphyrin		64.20%	N/A		
	COF-366-Cu-NS	Cu	Porphyrin		71%	N/A		
	Co-N-COF	Co	Others		97%	N/A		
HCOOH	Co-N-COF	Co	Others	Metal species and microstructures	97%	N/A	H-type cell and 1.0 M KHCO ₃	132
CH ₃ COOH	PcCu-TFPN	Cu	Phthalocyanine	Electronic modulation	90.3%	N/A	Flow cell and 0.1 M KHCO ₃	135

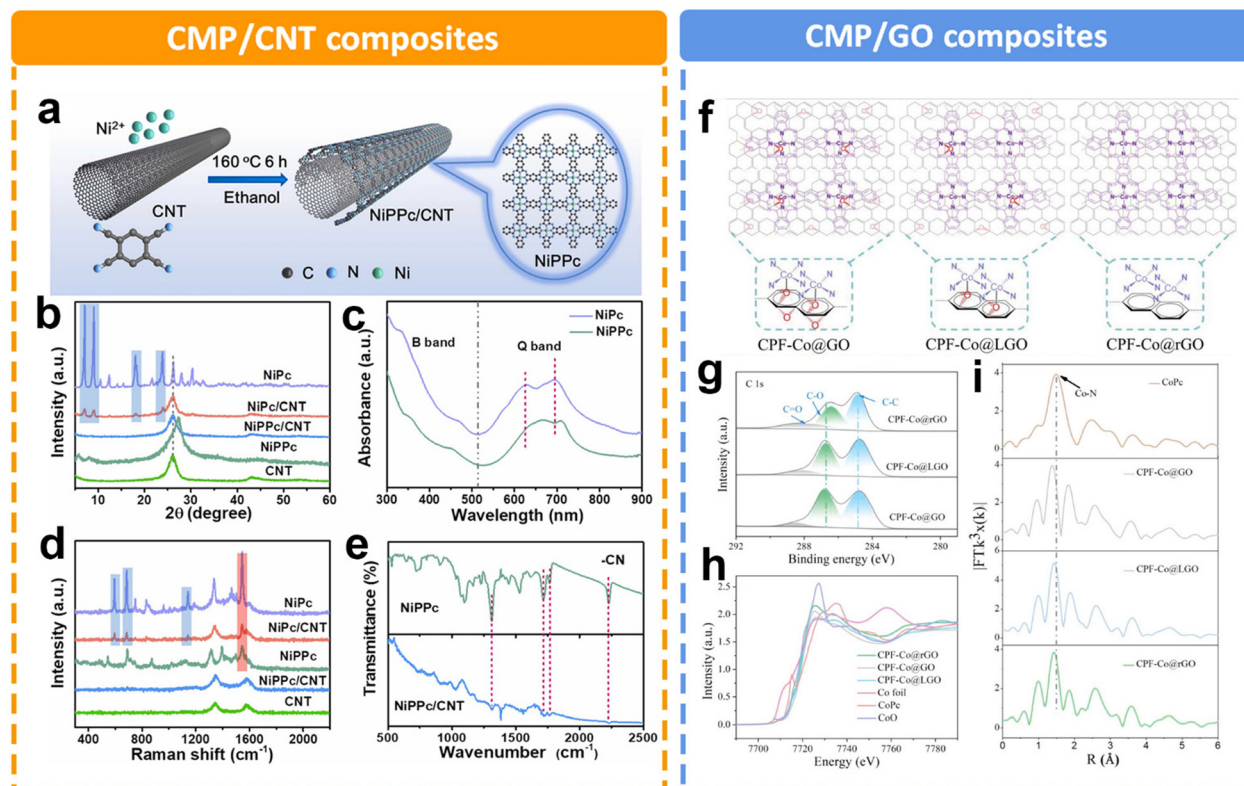


Fig. 13 Schematic illustration of different CMP composites and their characterization. (a) Schematic illustration of synthesizing NiPPc/CNTs, (b) XRD patterns of samples, (c) UV-Vis spectra of NiPc and NiPPc in DMF, (d) Raman spectra of samples, and (e) FTIR spectra of NiPPc and NiPPc/CNT.¹⁴⁶ (f) Schematic illustration of the CPF-Co@GO/LGO/rGO hybrid preparation and application to electrocatalytic CO₂ to CO conversion. Structural characterization of CPF-Co@GO, CPF-Co@LGO, and CPF-Co@rGO, (g) the high-resolution C 1s XPS spectra, (h) Co K edge XANES spectra, and (i) Co K-edge FT of the EXAFS spectra.¹⁴⁷



terns and Raman spectra indicate the successful synthesis of coaxial NiPPc/CNT (Fig. 13b and d). It also displays a red shift in Q-band absorption in contrast to NiPc, indicating NiPPc with an extended π conjugation (Fig. 13c). Electrochemical results show that NiPPc/CNT exhibits a high current density of $\sim 300 \text{ mA cm}^{-2}$ for CO_2RR with a CO selectivity of 99.8%. A similar research study was conducted by Zang *et al.*,¹⁴⁷ by introducing graphene oxide (GO) to modify the charge density distribution of CoPc (Fig. 13f). They designed and synthesized a conjugated phthalocyanine framework catalyst (CPF-Co) with abundant CoN_4 centers. Meanwhile, they further optimized the coordination microenvironment of Co species through the axial coordination between oxygen in the partially reduced GO (LGO) and the catalytic sites (Fig. 13g-i). The optimized O-CoN₄ structure could break the electron distribution symmetry of Co, thus reducing the energy barrier to the activation of CO_2 to COOH^* . By adjusting the content of oxygen, the appropriate

support could also facilitate the charge transfer efficiency between the matrix layer and the catalytic sites. The optimized CPF-Co@LGO exhibited a high CO selectivity (97.6%), TOF value (2.81 s^{-1}) and stability (24 h) at 21 mA cm^{-2} current density.

Other than the electronic modulation of MPc-containing CMPs from the support, another modulation strategy is the tuning of the metallic centers. The deliberately designed metal vacancies in Pc are proposed to influence the electrocatalytic performance of CMPs. Guo *et al.* adopted a solid-state ionothermal method to synthesize an ultrathin CMP sheath around CNTs by a Scholl coupling reaction of metal-free H_2Pc with CoPc in a controllable manner to regulate atomic dispersion of metal centers (Fig. 14a).¹⁴⁸ Copolymerization of H_2Pc and CoPc *via* the dehydrogenation of periphery benzene rings produces a highly cross-linked two-dimensional network. The incorporated H_2Pc unit serves not only as an electron

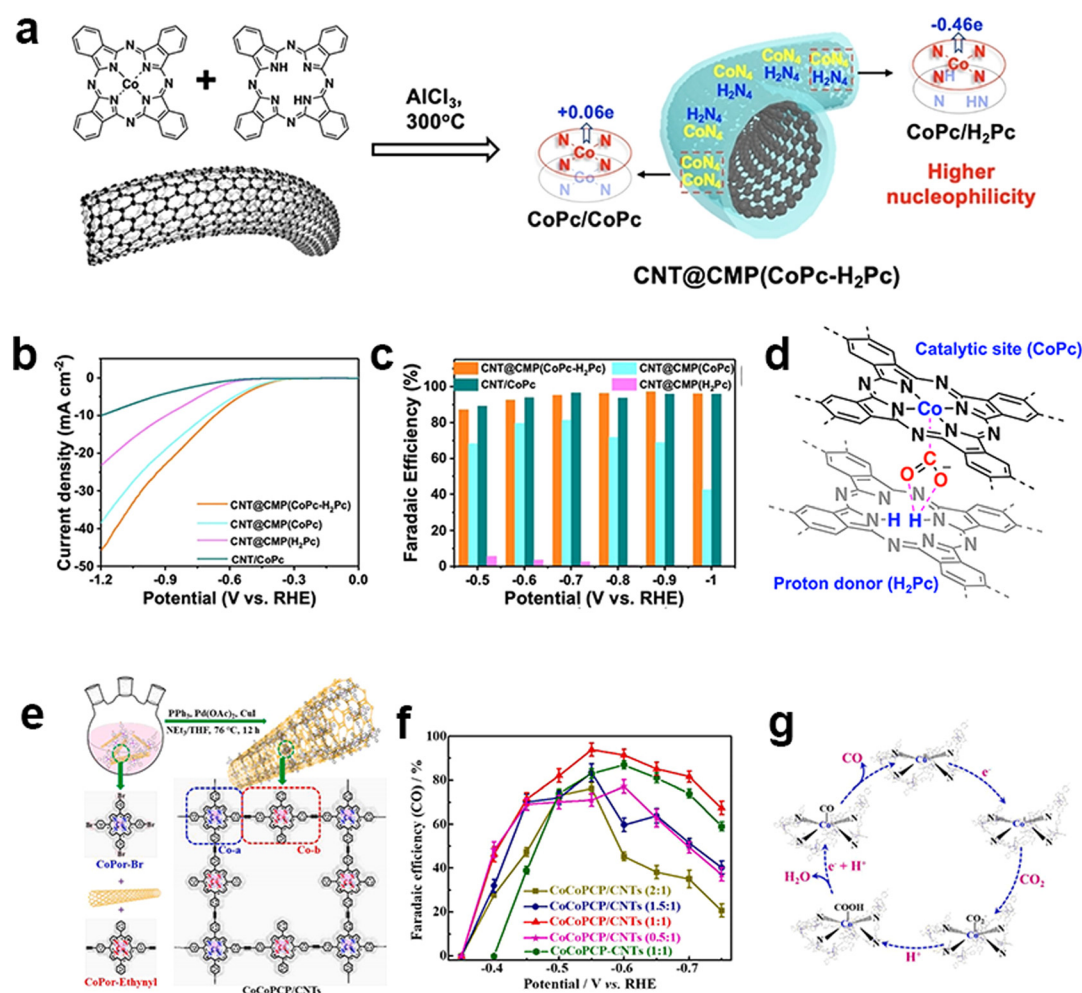


Fig. 14 Heterometallic design of CMPs. (a) Ionothermal synthesis of CNT@CMP(CoPc- H_2Pc) containing the stacked CoPc/CoPc and CoPc/ H_2Pc moieties with different Hirshfeld Co atomic charges, (b) polarization curves of CNT@CMP(CoPc- H_2Pc), CNT@CMP(CoPc) and CNT@CMP(H_2Pc) in CO_2 -saturated 0.5 M KHCO_3 solutions and (c) their CO FEs at different applied potentials, (d) proposed proton-donor mechanism of H_2Pc for the synergistic catalysis of CO_2 reduction.¹⁴⁸ (e) Schematic illustration of the *in situ* polymerization of CoCoPCP on CNTs by the Sonogashira coupling reaction, (f) CO faradaic efficiencies of the CoCoPCP/CNTs with different mass ratios and CoCoPCP-CNTs with a mass ratio of 1 : 1, and (g) the proposed mechanism (b) for the CO_2 RR process of CoCoPCP/CNTs.¹⁴⁹



donor improving the nucleophilicity of Co sites for CO₂ binding, but also as a proton donor to transport hydrogen to the Co active sites (Fig. 14d). The optimized catalyst generates a j_{CO} value of 15.2 mA cm⁻² (−0.6 V) and a high FE_{CO} (max. 97% at −0.9 V) in a broad potential window (Fig. 14b and c).

The incorporation of bimetallic sites is another plausible strategy to achieve electronic modulation through the synergistic effects between bimetallic sites. Peng *et al.* constructed ethynyl-linked Co-porphyrin based CMPs on CNTs to fabricate a bi-metallic central electrocatalyst (CoCoPCP/CNTs) for eCO₂RR (Fig. 14e).¹⁴⁹ It bears four-linked Co-porphyrin units (Co-a) and two-linked Co-porphyrin bridging ones (Co-b), which could promote the charge transfer and electronic modulation of Co active sites. It synergistically improves the electrocatalytic activity and selectivity of the eCO₂RR. The CoCoPCP/CNT composites have up to 94% FE_{CO} and excellent cycling stability (Fig. 14f). It could maintain a current density of −8.1 mA cm⁻² within 24 hours and the FE_{CO} could be achieved at more than 90% for a long time.

As most reported CMPs have a randomly packed morphology with highly stacking and aggregated structures, their active centers embedded in the cavities could not be fully exposed during the eCO₂RR. Reasonable nanostructural design is helpful to further improve their electrocatalytic efficiency and stability. Ye *et al.* reported an ultrathin CMP based on the squaraine-bridged cobalt tetraamino phthalocyanine structure (COP-SA) for high-performance eCO₂RR towards CO (Fig. 15a),¹⁵⁰ which has an extended π -conjugated skeleton with an ultrathin thickness of ~1.7 nm and an abundant microporous structure (Fig. 15c). The ultrathin nano-

structure avoids the agglomeration of molecular catalysts and is beneficial for the exposure of active sites. Theoretical calculation suggests that the squaraine unit will enhance *COOH adsorption and favor *CO desorption. These properties synergistically contribute to the high current density and FE_{CO} (Fig. 15f and g).

The main product of the eCO₂RR catalyzed by CMPs is CO, and other hydrocarbon products are rarely observed in previous research. Their production generally requires sufficient and highly active redox centers to realize the challenging proton-coupled electron transfer steps. Lan and his coworkers reported a facile and scalable strategy to prepare a series of Cu-porphyrin based CMPs with different linkages by *in situ* copolymerization of dialdehyde-based ligands and pyrrole.¹⁵¹ These catalysts show efficient electrocatalytic performance toward the electrocatalytic conversion of CO₂ to CH₄. The obtained materials are composed of directly bonded auxiliary groups and Cu porphyrin centers. Specifically, the optimized catalyst with ferrocene as a linker exhibited excellent CO₂-to-CH₄ electroreduction performance (current density, −491.5 mA cm⁻² and FE_{CH₄}, 75.9%). DFT calculations revealed that the directly interacting ferrocene groups enhance the electron-cloud density of Cu in the porphyrin-center. The strong adsorption ability of ferrocene groups toward OH* promotes the proton-coupled electron transfer reaction process, which kinetically improves the preferential CO₂-to-CH₄ pathway.

CMPs may have better intrinsic electric conductivity compared with COFs due to their well-conjugated structures. However, due to their short-range ordered and layer stacked structures, the obtained CMPs may have variable properties

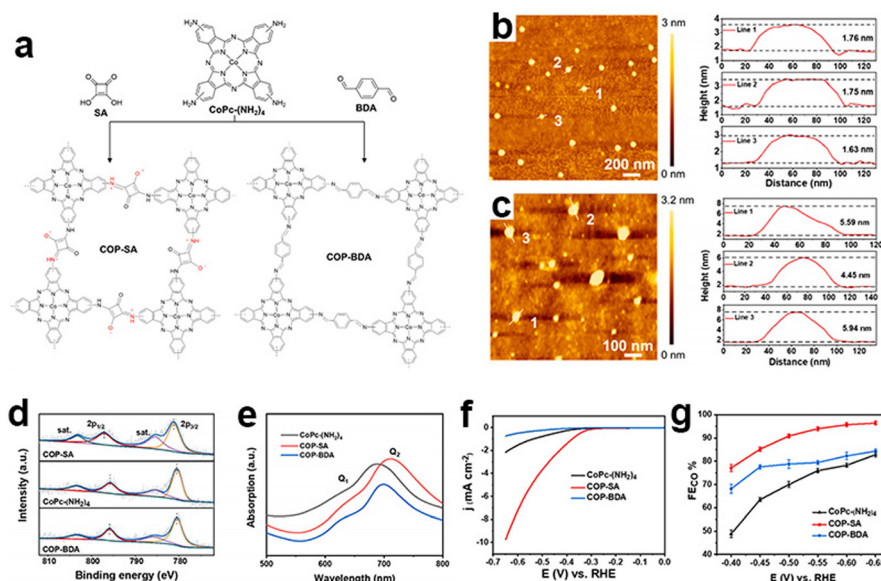


Fig. 15 Ultrathin nanostructured CMPs. (a) Schematic illustration of the synthesis of COP-SA and COP-BDA, (b) the atomic force microscopy topographical image of COP-SA and the height profile of lines 1, 2 and 3, (c) the atomic force microscopy topographical image of COP-BDA and the height profile of lines 1, 2 and 3, (d) high resolution Co(2p) XPS spectra and (e) UV-vis spectra of CoPc-(NH₂)₄, COP-SA and COP-BDA, (f) LSV curves of CoPc-(NH₂)₄, COP-SA and COP-BDA, and (g) faradaic efficiencies from −0.4 to −0.65 V vs. RHE of CoPc-(NH₂)₄, COP-SA and COP-BDA.¹⁵⁰



which are not as expected. This may bring about the challenge in designing high-performance CMP-based catalysts toward the eCO_2RR .

3.3 POCs

As discussed above, COFs and CMPs often have a large number of active sites blocked by interlayer stacking and steric hindrance, limiting their full utilization. In contrast, porous organic cages (POCs), which are discrete covalent-bonded molecular architectures, also known as covalent organic polyhedra, represent alternative promising candidates for CPCs. POCs possess the advantages of a large surface area, stable chemical properties and easy processing. Compared with COFs

and CMPs, POCs have fully exposed structures that maximize the active sites and speed up the mass transfer process. However, due to the spatial separated structures of POCs, the intrinsic conductivity is limited. Moreover, the synthesis of POCs is challenging. Within the past few years, only a few examples of POCs used in the eCO_2RR have been published, even though they have superiority in some aspects.

The preparation of POCs based on redoxative porphyrins has been an attractive research subject due to the accessible active sites, favorable guest diffusion and confinement inside the cage, and reduced aggregation of catalytic centers. Chang *et al.* proposed a method to increase the electrochemically active surface area and facilitate mass transport, which was



Fig. 16 Schematic illustration of different POCs and their performance. (a) Supramolecular structure of Fe-PB composed of six Fe-TPP monomer building blocks, (b) proposed model by which the permanently porous supramolecular Fe-PB facilitates the diffusion of the substrate and the electrolyte around the catalytic iron centers,¹⁵² (c) schematic synthesis of CPOC-M ($M = \text{H}_2$, Co(II) , Ni(II) , Cu(II) , and Zn(II)), (d) NPA charge of Co, Ni, Cu and Zn atoms in the porphyrin without the introduction of CO_2 ,¹⁵³ (e) the synthesis of ZnPc and NiPc cages, (f) partially blocked and readily accessible CO_2 reduction catalytic center in the aggregated NiPc-8OH monomer (left) and the NiPc cage (right), (g) LSV curve of ZnPc and NiPc cages in Ar- and CO_2 -saturated 0.1 M KHCO_3 solutions, respectively, (h) CO faradaic efficiency of ZnPc and NiPc cages, and (i) CO faradaic efficiency of the NiPc cage and NiPc-8OH.¹⁵⁴



Table 2 Typical CMPs and POCs as electrocatalysts in the eCO₂RR

Main product	Material	Metal center	Chelating sites	Design principle	Highest FE	TOF	Reaction conditions	Ref.
CO	COP-SA COP-BDA	Co	Phthalocyanine	Electronic modulation	96.50% 84.4%	46.3 s ⁻¹ 1.5 s ⁻¹	Home-made three-electrode electrochemical cell and 0.5 M KHCO ₃	150
CO	NiPcP	Ni	Phthalocyanine	Electronic modulation	~100%	23 148 h ⁻¹	Three-compartment flow cell and 1 M KOH	145
CO	NiPPc/CNT	Ni	Phthalocyanine	Electronic modulation	99.80%	N/A	Flow cell and 1 M KHCO ₃	146
CO	CoPPc/CNT	Co	Phthalocyanine	Electronic modulation	90%	4900 h ⁻¹	H type and 0.5 M NaHCO ₃	155
CO	CoCoPCP/CNTs CoH ₂ PCP/CNTs H ₂ CoPCP/CNTs	Co	Porphyrin	Electronic modulation and heterometallic centers	94% 85% 65%	2.4 s ⁻¹ N/A	H type cell and 0.5 M KHCO ₃	149
CO	CNT@CMP (CoPc-H ₂ Pc) CNT@CMP (CoPc) CNT@CMP (H ₂ Pc)	Co	Phthalocyanine	Electronic modulation and heterometallic centers	97% 82% 5%	97 592 h ⁻¹ N/A	Flow cell and 0.5 M KHCO ₃	148
CO	CPF-Co@GO CPF-Co@LGO CPF-Co@rGO	Metal free Co	Phthalocyanine	Electronic modulation	79.1% 97.6% 70.4%	1.07 s ⁻¹ 2.81 s ⁻¹ 2.27 s ⁻¹	Type cell and 0.5 M KHCO ₃	147
CH ₄	Fe-CPP-Cu Bz-CPP-Cu Bp-CPP-Cu Tp-CPP-Cu Fe-PB	Cu and Fe Cu	Porphyrin and cyclopentadienyl Porphyrin	Electronic modulation and heterometallic centers	75.90% 62.70% 45.80% 44.30%	0.013 s ⁻¹ N/A	Flow cell and 1 M KOH	151
CO	Fe-TPP	Fe	Porphyrin	Electronic modulation and microstructure	100 ± 2%	1.74 s ⁻¹	Custom-made three-piece glass electrochemical cell and 0.5 M KHCO ₃	152
CO	CPOC-Co	Co	Porphyrin	Metal species and microstructure	96 ± 3% 92%	0.94 s ⁻¹ 0.98 s ⁻¹	H-type cell and 0.5 M KHCO ₃	153
CO	NiPc cage ZnPc cage	Ni Zn	Phthalocyanine	Metal species and microstructure	98.7% <10%	606 h ⁻¹ N/A	Standard two-compartment electrochemical cell and 0.1 M KHCO ₃	154

installing iron tetraphenylporphyrin (Fe-TPP) into a rhombicuboctahedral POC for electrochemical CO₂-to-CO conversion (Fig. 16a).¹⁵² Fe-porphyrin based POCs could facilitate the diffusion of the substrate and electrolyte throughout an array of readily accessible catalytic metal centers (Fig. 16b). The porous architecture of POCs resulted in a larger amount of exposed electroactive iron centers compared with their monomeric counterpart Fe-TPP. The electrochemical test results showed enhanced stability and current density than Fe-TPP. Similarly, Jiang *et al.* conducted a comprehensive study of a functional cofacial porphyrin organic cage, CPOC-H₂, through the linking of a 5,10,15,20-tetrakis(4-formylphenyl)porphyrin and chiral (2-aminocyclohexyl)-1,4,5,8-naphthalenetetraformyl diimide (Fig. 16c).¹⁵³ The following post-synthetic metalation of the metal-free cage with different metal ions leads to the generation of four D-A-type cages. Among them, Co porphyrin based POCs have the most electron-rich environment with stronger physical adsorption capability for CO₂ molecules (Fig. 16d). According to the DFT calculation, the Co-C Mayer bond order (0.93) of the *COOH intermediate is the highest, indicating the key influence of metallic centers on the electrocatalytic performance toward the eCO₂RR. Co porphyrin based

POCs supply a partial current density of 18.0 mA cm⁻² at -0.9 V with the FE_{CO} of 90%.

In addition to porphyrin-based POCs, phthalocyanine-based POCs for eCO₂RR have been developed recently in our group.¹⁵⁴ Two kinds of well-defined phthalocyanine-based organic cages chelated with Ni and Zn ions were obtained *via* one-step dynamic spiroborate formation in high yields for the first time (Fig. 16e). The site-isolated redox-active metal centers, readily accessible intrinsic cavity, and the shape persistency of the cage structure precludes the aggregation of adjacent NiPc panels, which maintains the number of electrochemically accessible and active Pc centers (Fig. 16f). The Ni-metalated Pc (NiPc) cage exhibits high catalytic efficiency, selectivity, and stability toward the eCO₂RR. LSV results show that the onset potential of the NiPc cage is -0.60 V, which is better than that of the ZnPc cage (-0.74 V) (Fig. 16g). It almost exclusively generates CO with a selectivity of 98.7% at -1.2 V (Fig. 16h). The selectivity is maintained at over 95% in long-term durability tests, which demonstrates the advantages of integrating electrocatalytically active NiPc moieties into a molecular cage structure (Fig. 16i). Additionally, due to the structural superiority of POCs, they could also be applied as



additives to modify the electrocatalytic eCO_2RR of other catalysts, allowing the facile contact of CO_2 molecules with the active sites. This research will not be discussed in this review.

POCs exhibit the most well-defined catalyst structure but low intrinsic conductivity. It is assumed that well-defined POCs could have their active sites fully exposed during electrocatalysis. However, electrons are difficult to transfer directly between different POC molecules. Typically, incorporating POCs with conductive materials are needed. How to ensure their uniform and non-aggregated distribution on conductive materials is still challenging, which may be one of the key problems in developing efficient POC-based electrocatalysts for eCO_2RR .

Typical CMPs and POCs as electrocatalysts in the eCO_2RR are summarized in Table 2.

4. Conclusion and perspectives

Covalent porous catalysts (CPCs) are considered promising catalysts for eCO_2RR due to their well-defined structures, high specific surface areas and high atomic utilization. Moreover, their well-defined chemical structures at the molecular level provide excellent platforms for studying the catalytic mechanism for eCO_2RR , which could guide the rational design of high-performance eCO_2RR catalysts. In this review, we summarized the design principle and specific application in the eCO_2RR of three typical covalent porous catalysts (COFs, CMPs and organic cages). We focus on the CPCs' intrinsic activity enhancing methods including the design of metal centers, chelating groups, organic backbones, and side chain groups. It could be concluded that the key factors through these modulations lie in the electronic properties of the active metallic sites embedded in the porous structure and the porosity of the organic backbone. The electronic properties of the active metallic sites mainly determine the activity and selectivity, while the porosity of the organic backbone mainly determines the efficiency of CPCs. Moreover, compounded CPCs with carbon additives, such as CNTs and graphene, facilitate the electron transfer during the catalytic process. With the development of synthetic methods and suitable functional building blocks, a number of CPCs have shown remarkable eCO_2RR performance in previous research. Some challenges still stand, which need further exploration.

(1) Accurate relationship between CPC structures and eCO_2RR performance. Due to the two-dimensional stacking structures of 2D COFs and CMPs, the active sites in the adjacent layers may be interfered by each other, which could alter or reduce their certain performance when compared with independent single metal sites. This problem is overlooked in current works when exploring the relationship between CPC structures and eCO_2RR performance. It is expected that CPCs with clearer structures, combined with more advanced characterization methods and theoretical calculation, will help gain a better understanding. Moreover, multiscale theoretical simulations, such as finite element simulation, quantum mech-

anics, and machine learning, are needed to better understand the influences of the porosity of the catalyst and mass transfer on the reaction process.

(2) The large-scale practical application of CPCs for eCO_2RR . The current density catalysed by CPCs during the eCO_2RR still falls far short of meeting practical requirements due to the slow reaction kinetics and lower intrinsic electrical conductivity of CPCs. Due to the organic nature of CPC backbones, the conductivity of CPCs is relatively low, when compared with metal-based electrocatalysts. Low conductivity may cause high resistance and low efficiency during the eCO_2RR . Moreover, due to the coordinated metal leaching and conjugated structure destruction, the stability of CPCs in the long-term operation also represents a challenge. Designing advanced reactors or combining CPCs with conductive substrates can significantly improve the catalytic efficiency of CPCs for eCO_2RR . Enhancing the metal coordination strength and improving the robustness of the linkages between building blocks (such as sp^2 carbon conjugated and cyclized linkages) are highly needed, which may improve the stability of CPCs for eCO_2RR . Additionally, to realize large-scale synthesis, the high cost of monomers used for the construction of CPCs becomes another obstacle in their practical application. Therefore, an optimized and low-cost industrial supply of various monomers is also urgently needed in the future.

(3) The deep reduction pathway of the eCO_2RR on CPCs with different metal centers. Although the deep reduction of CO_2 to HCHO , CH_3OH and CH_4 has been reported on single atom catalysts besides Cu species, it is still difficult to achieve it on CPCs. Only several products, such as CO and HCOOH (2e^- transfer products) and CH_4 (8e^- transfer products), have been reported on CPCs. The important semi-hydrogenated products, such as HCHO and CH_3OH , have not been obtained by eCO_2RR on CPCs at present, which is worth paying attention. The reduction degree is determined by the binding strength of the key intermediates on the active sites. We suspect that rational modulation of the charge density on metal centers may optimize the adsorption of key intermediates for deep reduction, even facilitating C–C coupling to obtain C_{2+} products.

(4) Real active sites on CPCs for eCO_2RR . In the catalytic process of the eCO_2RR , it has been observed that some metal species can transform into metal clusters, which serve as the actual active sites for the reaction. However, many studies have previously presumed that the coordinated metal centers are the active sites, leading to a divergence between the experimental results and analysis. To bridge this gap, it is essential to employ advanced analytical techniques that can monitor the evolution of metal species (including metal bonds, coordination bonds and valence states). By doing so, a more accurate understanding of the genuine active sites responsible for eCO_2RR catalysis by CPCs can be achieved.

Conflicts of interest

There are no conflicts to declare.



Acknowledgements

The authors would like to thank the National Natural Science Foundation of China (NSFC) (Grants No. 21905115, 52173201), the China Postdoctoral Science Foundation (Grants No. 2020M671334), the MOE & SAFEA, 111 Project (B13025), the National First-Class Discipline Program of Light Industry Technology and Engineering (LITE2018-19), the Fundamental Research Funds for the Central Universities (JUSRP11929), and the Postgraduate Research & Practice Innovation Program of Jiangsu Province (KYCX23_2556). W. Z. thanks the University of Colorado Boulder for financial support.

References

- 1 CenCO2PIP, *Science*, 2023, **382**, eadi5177.
- 2 J. M. Chen, *Innovation*, 2021, **2**, 100127.
- 3 L. Chen, G. Msigwa, M. Yang, A. I. Osman, S. Fawzy, D. W. Rooney and P. S. Yap, *Environ. Chem. Lett.*, 2022, **20**, 2277–2310.
- 4 S. A. Montzka, E. J. Dlugokencky and J. H. Butler, *Nature*, 2011, **476**, 43–50.
- 5 M. R. Shaw, E. S. Zavaleta, N. R. Chiariello, E. E. Cleland, H. A. Mooney and C. B. Field, *Science*, 2002, **298**, 1987–1990.
- 6 B. Grignard, S. Gennen, C. Jerome, A. W. Kleij and C. Detrembleur, *Chem. Soc. Rev.*, 2019, **48**, 4466–4514.
- 7 G. Wang, J. Chen, Y. Ding, P. Cai, L. Yi, Y. Li, C. Tu, Y. Hou, Z. Wen and L. Dai, *Chem. Soc. Rev.*, 2021, **50**, 4993–5061.
- 8 L. Li, X. Li, Y. Sun and Y. Xie, *Chem. Soc. Rev.*, 2022, **51**, 1234–1252.
- 9 K. S. Song, P. W. Fritz and A. Coskun, *Chem. Soc. Rev.*, 2022, **51**, 9831–9852.
- 10 X. Jiang, X. Nie, X. Guo, C. Song and J. G. Chen, *Chem. Rev.*, 2020, **120**, 7984–8034.
- 11 C. Hepburn, E. Adlen, J. Beddington, E. A. Carter, S. Fuss, N. M. Dowell, J. C. Minx, P. Smith and C. K. Williams, *Nature*, 2019, **575**, 87–97.
- 12 F. Julia-Hernandez, T. Moragas, J. Cornella and R. Martin, *Nature*, 2017, **545**, 84–88.
- 13 A. Banerjee, G. R. Dick, T. Yoshino and M. W. Kanan, *Nature*, 2016, **531**, 215–219.
- 14 K. Sekine and T. Yamada, *Chem. Soc. Rev.*, 2016, **45**, 4524–4532.
- 15 H. Rao, L. C. Schmidt, J. Bonin and M. Robert, *Nature*, 2017, **548**, 74–77.
- 16 T. Kong, Y. Jiang and Y. Xiong, *Chem. Soc. Rev.*, 2020, **49**, 6579–6591.
- 17 W. Kim, B. A. McClure, E. Edri and H. Frei, *Chem. Soc. Rev.*, 2016, **45**, 3221–3243.
- 18 S. Navarro-Jaen, M. Virginie, J. Bonin, M. Robert, R. Wojcieszak and A. Y. Khodakov, *Nat. Rev. Chem.*, 2021, **5**, 564–579.
- 19 S. Fang, M. Rahaman, J. Bharti, E. Reisner, M. Robert, G. A. Ozin and Y. H. Hu, *Nat. Rev. Methods Primers*, 2023, **3**, 61.
- 20 F. Li, A. Thevenon, A. Rosas-Hernandez, Z. Wang, Y. Li, C. M. Gabardo, A. Ozden, C. T. Dinh, J. Li, Y. Wang, J. P. Edwards, Y. Xu, C. McCallum, L. Tao, Z. Q. Liang, M. Luo, X. Wang, H. Li, C. P. O'Brien, C. S. Tan, D. H. Nam, R. Quintero-Bermudez, T. T. Zhuang, Y. C. Li, Z. Han, R. D. Britt, D. Sinton, T. Agapie, J. C. Peters and E. H. Sargent, *Nature*, 2020, **577**, 509–513.
- 21 B. A. Rosen, A. Salehi-Khojin, M. R. Thorson, W. Zhu, D. T. Whipple, P. J. A. Kenis and R. I. Masel, *Science*, 2011, **334**, 643–644.
- 22 Y. Chen, X. Y. Li, Z. Chen, A. Ozden, J. E. Huang, P. Ou, J. Dong, J. Zhang, C. Tian, B. H. Lee, X. Wang, S. Liu, Q. Qu, S. Wang, Y. Xu, R. K. Miao, Y. Zhao, Y. Liu, C. Qiu, J. Abed, H. Liu, H. Shin, D. Wang, Y. Li, D. Sinton and E. H. Sargent, *Nat. Nanotechnol.*, 2024, **19**, 311–318.
- 23 P. P. Yang and M. R. Gao, *Chem. Soc. Rev.*, 2023, **52**, 4343–4380.
- 24 S. Gleizer, R. Ben-Nissan, Y. M. Bar-On, N. Antonovsky, E. Noor, Y. Zohar, G. Jona, E. Krieger, M. Shamshoum, A. Bar-Even and R. Milo, *Cell*, 2019, **179**, 1255–1263.
- 25 E. W. Yemm and A. J. Willis, *Nature*, 1954, **173**, 726–726.
- 26 T. E. Miller, T. Beneyton, T. Schwander, C. Diehl, M. Girault, R. McLean, T. Chotel, P. Claus, N. S. Cortina, J.-C. Baret and T. J. Erb, *Science*, 2020, **368**, 649–654.
- 27 A. Burlacot, O. Dao, P. Auroy, S. Cuine, Y. Li-Beisson and G. Peltier, *Nature*, 2022, **605**, 366–371.
- 28 Y. Y. Birdja, E. Pérez-Gallent, M. C. Figueiredo, A. J. Göttele, F. Calle-Vallejo and M. T. M. Koper, *Nat. Energy*, 2019, **4**, 732–745.
- 29 L. Zhang, Z. J. Zhao, T. Wang and J. Gong, *Chem. Soc. Rev.*, 2018, **47**, 5423–5443.
- 30 E. E. Benson, C. P. Kubiak, A. J. Sathrum and J. M. Smieja, *Chem. Soc. Rev.*, 2009, **38**, 89–99.
- 31 K. Zhao and X. Quan, *ACS Catal.*, 2021, **11**, 2076–2097.
- 32 S. Nitopi, E. Bertheussen, S. B. Scott, X. Liu, A. K. Engstfeld, S. Horch, B. Seger, I. E. L. Stephens, K. Chan, C. Hahn, J. K. Nørskov, T. F. Jaramillo and I. Chorkendorff, *Chem. Rev.*, 2019, **119**, 7610–7672.
- 33 H.-Q. Liang, S. Zhao, X.-M. Hu, M. Ceccato, T. Skrydstrup and K. Daasbjerg, *ACS Catal.*, 2021, **11**, 958–966.
- 34 L. Lin, T. Liu, J. Xiao, H. Li, P. Wei, D. Gao, B. Nan, R. Si, G. Wang and X. Bao, *Angew. Chem., Int. Ed.*, 2020, **59**, 22408–22413.
- 35 X. Zheng, P. De Luna, F. P. García de Arquer, B. Zhang, N. Becknell, M. B. Ross, Y. Li, M. N. Banis, Y. Li, M. Liu, O. Voznyy, C. T. Dinh, T. Zhuang, P. Stadler, Y. Cui, X. Du, P. Yang and E. H. Sargent, *Joule*, 2017, **1**, 794–805.
- 36 Y. Wu, Z. Jiang, X. Lu, Y. Liang and H. Wang, *Nature*, 2019, **575**, 639–642.
- 37 C. Choi, X. Wang, S. Kwon, J. L. Hart, C. L. Rooney, N. J. Harmon, Q. P. Sam, J. J. Cha, W. A. Goddard III, M. Elimelech and H. Wang, *Nat. Nanotechnol.*, 2023, **18**, 160–167.



- 38 L. Xu, J. Feng, L. Wu, X. Song, X. Tan, L. Zhang, X. Ma, S. Jia, J. Du, A. Chen, X. Sun and B. Han, *Green Chem.*, 2023, **25**, 1326–1331.
- 39 H. Chen, C. Ding, C. Kang, J. Zeng, Y. Li, Y. Li, Y. Li, C. Li and J. He, *Green Chem.*, 2023, **25**, 5320–5337.
- 40 A. Goyal, G. Marcandalli, V. A. Mints and M. T. M. Koper, *J. Am. Chem. Soc.*, 2020, **142**, 4154–4161.
- 41 P. Yang, Z. J. Zhao, X. Chang, R. Mu, S. Zha, G. Zhang and J. Gong, *Angew. Chem., Int. Ed.*, 2018, **57**, 7724–7728.
- 42 C. Kim, F. Dionigi, V. Beermann, X. Wang, T. Moller and P. Strasser, *Adv. Mater.*, 2019, **31**, 1805617.
- 43 J. Hao, Z. Zhuang, J. Hao, K. Cao, Y. Hu, W. Wu, S. Lu, C. Wang, N. Zhang, D. Wang, M. Du and H. Zhu, *ACS Nano*, 2022, **16**, 3251–3263.
- 44 Y. Jiang, J. Shan, P. Wang, L. Huang, Y. Zheng and S.-Z. Qiao, *ACS Catal.*, 2023, **13**, 3101–3108.
- 45 J. Wang, S. Kattel, C. J. Hawxhurst, J. H. Lee, B. M. Tackett, K. Chang, N. Rui, C. J. Liu and J. G. Chen, *Angew. Chem., Int. Ed.*, 2019, **58**, 6271–6275.
- 46 X. Yang, J. H. Lee, S. Kattel, B. Xu and J. G. Chen, *Nano Lett.*, 2022, **22**, 4576–4582.
- 47 L. Liu, F. Wang, X. Chu, L. Zhang, S. Zhang, X. Wang, G. Che, S. Song and H. Zhang, *Adv. Energy Mater.*, 2024, **14**, 2301575.
- 48 R. Zhao, Y. Wang, G. Ji, J. Zhong, F. Zhang, M. Chen, S. Tong, P. Wang, Z. Wu, B. Han and Z. Liu, *Adv. Mater.*, 2023, **35**, 2205262.
- 49 O. Christensen, S. Zhao, Z. Sun, A. Bagger, J. V. Lauritsen, S. U. Pedersen, K. Daasbjerg and J. Rossmeisl, *ACS Catal.*, 2022, **12**, 15737–15749.
- 50 Y. Zhang, P. Li, C. Zhao, G. Zhou, F. Zhou, Q. Zhang, C. Su and Y. Wu, *Sci. Bull.*, 2022, **67**, 1679–1687.
- 51 Z. Ma, T. Wan, D. Zhang, J. A. Yuwono, C. Tsounis, J. Jiang, Y. H. Chou, X. Lu, P. V. Kumar, Y. H. Ng, D. Chu, C. Y. Toe, Z. Han and R. Amal, *ACS Nano*, 2023, **17**, 2387–2398.
- 52 Q. Zhang and J. Guan, *Adv. Funct. Mater.*, 2020, **30**, 2000768.
- 53 M. Li, H. Wang, W. Luo, P. C. Sherrell, J. Chen and J. Yang, *Adv. Mater.*, 2020, **32**, 2001848.
- 54 F. Yang, P. Song, X. Liu, B. Mei, W. Xing, Z. Jiang, L. Gu and W. Xu, *Angew. Chem., Int. Ed.*, 2018, **57**, 12303–12307.
- 55 Y. Hou, Y.-B. Huang, Y.-L. Liang, G.-L. Chai, J.-D. Yi, T. Zhang, K.-T. Zang, J. Luo, R. Xu, H. Lin, S.-Y. Zhang, H.-M. Wang and R. Cao, *CCS Chem.*, 2019, **1**, 384–395.
- 56 L. Lin, Q. Zhang, Y. Ni, L. Shang, X. Zhang, Z. Yan, Q. Zhao and J. Chen, *Chem*, 2022, **8**, 1822–1854.
- 57 K. Wang, D. Qi, Y. Li, T. Wang, H. Liu and J. Jiang, *Coord. Chem. Rev.*, 2019, **378**, 188–206.
- 58 C. Wang, Z. Lv, W. Yang, X. Feng and B. Wang, *Chem. Soc. Rev.*, 2023, **52**, 1382–1427.
- 59 J. M. Lee and A. I. Cooper, *Chem. Rev.*, 2020, **120**, 2171–2214.
- 60 F. Franco, C. Rettenmaier, H. S. Jeon and B. R. Cuenya, *Chem. Soc. Rev.*, 2020, **49**, 6884–6946.
- 61 R. Freund, O. Zaremba, G. Arnauts, R. Ameloot, G. Skorupskii, M. Dinca, A. Bavykina, J. Gascon, A. Ejsmont, J. Goscińska, M. Kalmutzki, U. Lachelt, E. Ploetz, C. S. Diercks and S. Wuttke, *Angew. Chem., Int. Ed.*, 2021, **60**, 23975–24001.
- 62 J. Li, X. Jing, Q. Li, S. Li, X. Gao, X. Feng and B. Wang, *Chem. Soc. Rev.*, 2020, **49**, 3565–3604.
- 63 Y. Yusran, Q. Fang and V. Valtchev, *Adv. Mater.*, 2020, **32**, 2002038.
- 64 Q. Guan, L.-L. Zhou and Y.-B. Dong, *Chem. Soc. Rev.*, 2022, **51**, 6307–6416.
- 65 X. Li, P. Yadav and K. P. Loh, *Chem. Soc. Rev.*, 2020, **49**, 4835–4866.
- 66 Z. Li, T. He, Y. Gong and D. Jiang, *Acc. Chem. Res.*, 2020, **53**, 1672–1685.
- 67 H. Ma, Y. Chen, X. Li and B. Li, *Adv. Funct. Mater.*, 2021, **31**, 2101861.
- 68 X. Yang, Z. Ullah, J. F. Stoddart and C. T. Yavuz, *Chem. Rev.*, 2023, **123**, 4602–4634.
- 69 R. D. Mukhopadhyay, Y. Kim, J. Koo and K. Kim, *Acc. Chem. Res.*, 2018, **51**, 2730–2738.
- 70 H. Wang, Y. Jin, N. Sun, W. Zhang and J. Jiang, *Chem. Soc. Rev.*, 2021, **50**, 8874–8886.
- 71 J. Dong, X. Han, Y. Liu, H. Li and Y. Cui, *Angew. Chem., Int. Ed.*, 2020, **59**, 13722–13733.
- 72 Z. Li, X. Zhang, H. Cheng, J. Liu, M. Shao, M. Wei, D. G. Evans, H. Zhang and X. Duan, *Adv. Energy Mater.*, 2019, **10**, 1900486.
- 73 Y. Cao, W. Peng, Y. Li, F. Zhang, Y. Zhu and X. Fan, *Green Energy Environ.*, 2023, **8**, 360–382.
- 74 Q. Pan, Z. Lei, Y. Zhao and W. Zhang, *EnergyChem*, 2023, **5**, 100107.
- 75 L. Xiao, L. Qi, J. Sun, A. Husile, S. Zhang, Z. Wang and J. Guan, *Nano Energy*, 2024, **120**, 109155.
- 76 S. Suleman, Y. Zhang, Y. Qian, J. Zhang, Z. Lin, O. Metin, Z. Meng and H. L. Jiang, *Angew. Chem., Int. Ed.*, 2024, **63**, 202314988.
- 77 Y. Mou, X. Wu, C. Qin, J. Chen, Y. Zhao, L. Jiang, C. Zhang, X. Yuan, E. H. Ang and H. Wang, *Angew. Chem., Int. Ed.*, 2023, **62**, 202309480.
- 78 S. Daliran, A. R. Oveisi, Y. Peng, A. López-Magano, M. Khajeh, R. Mas-Ballesté, J. Alemán, R. Luque and H. Garcia, *Chem. Soc. Rev.*, 2022, **51**, 7810–7882.
- 79 P. De La Torre, L. An and C. J. Chang, *Adv. Mater.*, 2023, **35**, 2302122.
- 80 A. Wagner, C. D. Sahm and E. Reisner, *Nat. Catal.*, 2020, **3**, 775–786.
- 81 J. Xu, Y. Xu and X. H. Bu, *Small*, 2021, **17**, 2102331.
- 82 Z. Liang, H. Y. Wang, H. Zheng, W. Zhang and R. Cao, *Chem. Soc. Rev.*, 2021, **50**, 2540–2581.
- 83 X. Cui, S. Lei, A. C. Wang, L. Gao, Q. Zhang, Y. Yang and Z. Lin, *Nano Energy*, 2020, **70**, 104525.
- 84 D. H. Yang, Y. Tao, X. Ding and B. H. Han, *Chem. Soc. Rev.*, 2022, **51**, 761–791.
- 85 P. Xiao and Y. Xu, *J. Mater. Chem. A*, 2018, **6**, 21676–21695.



- 86 M. K. Lee, M. Shokouhimehr, S. Y. Kim and H. W. Jang, *Adv. Energy Mater.*, 2022, **12**, 2003990.
- 87 X. Cui, M. Wu, X. Liu, B. He, Y. Zhu, Y. Jiang and Y. Yang, *Chem. Soc. Rev.*, 2024, **53**, 1447–1494.
- 88 W. Ma, X. He, W. Wang, S. Xie, Q. Zhang and Y. Wang, *Chem. Soc. Rev.*, 2021, **50**, 12897–12914.
- 89 F.-Y. Gao, R.-C. Bao, M.-R. Gao and S.-H. Yu, *J. Mater. Chem. A*, 2020, **8**, 15458–15478.
- 90 X. Tan, C. Yu, Y. Ren, S. Cui, W. Li and J. Qiu, *Energy Environ. Sci.*, 2021, **14**, 765–780.
- 91 J. J. Lv, R. Yin, L. Zhou, J. Li, R. Kikas, T. Xu, Z. J. Wang, H. Jin, X. Wang and S. Wang, *Angew. Chem., Int. Ed.*, 2022, **61**, 202207252.
- 92 Z. Wang, Y. Zhou, P. Qiu, C. Xia, W. Fang, J. Jin, L. Huang, P. Deng, Y. Su, R. Crespo-Otero, X. Tian, B. You, W. Guo, D. Di Tommaso, Y. Pang, S. Ding and B. Y. Xia, *Adv. Mater.*, 2023, **35**, 2303052.
- 93 R. He, N. Xu, I. M. u. Hasan, L. Peng, L. Li, H. Huang and J. Qiao, *EcoMat*, 2023, **5**, 12346.
- 94 Y. Kuang, H. Rabiee, L. Ge, T. E. Rufford, Z. Yuan, J. Bell and H. Wang, *Energy Environ. Sci.*, 2023, **6**, 12596.
- 95 Y. Li, H. Wang, X. Yang, T. O'Carroll and G. Wu, *Angew. Chem., Int. Ed.*, 2024, e202317884.
- 96 L. Zhang, N. Jin, Y. Yang, X. Y. Miao, H. Wang, J. Luo and L. Han, *Nano-Micro Lett.*, 2023, **15**, 228.
- 97 K. E. R. Cruz, Y. Liu, T. L. Soucy, P. M. Zimmerman and C. C. L. McCrory, *ACS Catal.*, 2021, **11**, 13203–13216.
- 98 S. M. Lee, W. S. Cheon, M. G. Lee and H. W. Jang, *Small Struct.*, 2022, **4**, 2200236.
- 99 J. Wang, Y. Song, C. Chen, X. Zhao and W. Fan, *ACS Catal.*, 2023, **13**, 15794–15810.
- 100 H. Wang, Y. Tong and P. Chen, *Nano Energy*, 2023, **118**, 108967.
- 101 C. Hu, Y. Zhang, A. Hu, Y. Wang, X. Wei, K. Shen, L. Chen and Y. Li, *Adv. Mater.*, 2023, **35**, 2209298.
- 102 S. Wei, W. Liu, C. Yang, P. Bai, X. Kong, W. Sun and L. Xu, *Mater. Chem. Front.*, 2023, **7**, 4723–4743.
- 103 T. Tang, Z. Wang and J. Guan, *Adv. Funct. Mater.*, 2022, **32**, 2111504.
- 104 H. Chen, X. Liang, Y. Liu, X. Ai, T. Asefa and X. Zou, *Adv. Mater.*, 2020, **32**, 2002435.
- 105 H. Li, F. Pan, C. Qin, T. Wang and K. J. Chen, *Adv. Energy Mater.*, 2023, **13**, 2301378.
- 106 S. Lin, C. S. Diercks, Y.-B. Zhang, N. Kornienko, E. M. Nichols, Y. Zhao, A. R. Paris, D. Kim, P. Yang, O. M. Yaghi, C. J. Chang, *et al.*, *Science*, 2015, **349**, 1208–1213.
- 107 H. Dong, M. Lu, Y. Wang, H.-L. Tang, D. Wu, X. Sun and F.-M. Zhang, *Appl. Catal., B*, 2022, **303**, 120897.
- 108 G. C. D. Bando, S. S. Mondal, F. Franco, A. Bucci, V. Martin-Diaconescu, M. A. Ortuño, P. H. van Langevelde, A. Shafir, N. López and J. Lloret-Fillol, *ACS Catal.*, 2021, **11**, 7210–7222.
- 109 P. L. Cheung, S. K. Lee and C. P. Kubiak, *Chem. Mater.*, 2019, **31**, 1908–1919.
- 110 Q. Wu, R.-K. Xie, M.-J. Mao, G.-L. Chai, J.-D. Yi, S.-S. Zhao, Y.-B. Huang and R. Cao, *ACS Energy Lett.*, 2020, **5**, 1005–1012.
- 111 H.-J. Zhu, M. Lu, Y.-R. Wang, S.-J. Yao, M. Zhang, Y.-H. Kan, J. Liu, Y. Chen, S.-L. Li and Y.-Q. Lan, *Nat. Commun.*, 2020, **11**, 497.
- 112 Q. Wu, M. J. Mao, Q. J. Wu, J. Liang, Y. B. Huang and R. Cao, *Small*, 2020, **17**, 2004933.
- 113 X. Zhang, Y.-Z. Yuan, H.-F. Li, Q.-J. Wu, H.-J. Zhu, Y.-L. Dong, Q. Wu, Y.-B. Huang and R. Cao, *Mater. Chem. Front.*, 2023, **7**, 2661–2670.
- 114 S. An, C. Lu, Q. Xu, C. Lian, C. Peng, J. Hu, X. Zhuang and H. Liu, *ACS Energy Lett.*, 2021, **6**, 3496–3502.
- 115 X. Yang, X. Li, M. Liu, S. Yang, Q. Niu, L. Zhai, Z. Jiang, Q. Xu and G. Zeng, *ACS Mater. Lett.*, 2023, **5**, 1611–1618.
- 116 M. Liu, S. Yang, X. Yang, C.-X. Cui, G. Liu, X. Li, J. He, G. Z. Chen, Q. Xu and G. Zeng, *Nat. Commun.*, 2023, **14**, 3800.
- 117 C. S. Diercks, S. Lin, N. Kornienko, E. A. Kapustin, E. M. Nichols, C. Zhu, Y. Zhao, C. J. Chang and O. M. Yaghi, *J. Am. Chem. Soc.*, 2018, **140**, 1116–1122.
- 118 Y. Lu, J. Zhang, W. Wei, D.-D. Ma, X.-T. Wu and Q.-L. Zhu, *ACS Appl. Mater. Interfaces*, 2020, **12**, 37986–37992.
- 119 T. He, C. Yang, Y. Chen, N. Huang, S. Duan, Z. Zhang, W. Hu and D. Jiang, *Adv. Mater.*, 2022, **34**, 2205186.
- 120 N. Huang, K. H. Lee, Y. Yue, X. Xu, S. Irle, Q. Jiang and D. Jiang, *Angew. Chem., Int. Ed.*, 2020, **59**, 16587–16593.
- 121 M. D. Zhang, D. H. Si, J. D. Yi, S. S. Zhao, Y. B. Huang and R. Cao, *Small*, 2020, **16**, 2005254.
- 122 N. Huang, K. H. Lee, Y. Yue, X. Xu, S. Irle, Q. Jiang and D. Jiang, *Angew. Chem., Int. Ed.*, 2020, **59**, 16587–16593.
- 123 C. L. Yao, J. C. Li, W. Gao and Q. Jiang, *Chem. – Eur. J.*, 2018, **24**, 11051–11058.
- 124 B. Han, X. Ding, B. Yu, H. Wu, W. Zhou, W. Liu, C. Wei, B. Chen, D. Qi, H. Wang, K. Wang, Y. Chen, B. Chen and J. Jiang, *J. Am. Chem. Soc.*, 2021, **143**, 7104–7113.
- 125 Y. Yue, P. Cai, K. Xu, H. Li, H. Chen, H.-C. Zhou and N. Huang, *J. Am. Chem. Soc.*, 2021, **143**, 18052–18060.
- 126 J. Yuan, S. Chen, Y. Zhang, R. Li, J. Zhang and T. Peng, *Adv. Mater.*, 2022, **34**, 2203139.
- 127 L. Gong, Y. Gao, Y. Wang, B. Chen, B. Yu, W. Liu, B. Han, C. Lin, Y. Bian, D. Qi and J. Jiang, *Catal. Sci. Technol.*, 2022, **12**, 6566–6571.
- 128 S.-Y. Chi, Q. Chen, S.-S. Zhao, D.-H. Si, Q.-J. Wu, Y.-B. Huang and R. Cao, *J. Mater. Chem. A*, 2022, **10**, 4653–4659.
- 129 B. Han, Y. Jin, B. Chen, W. Zhou, B. Yu, C. Wei, H. Wang, K. Wang, Y. Chen, B. Chen and J. Jiang, *Angew. Chem., Int. Ed.*, 2022, **61**, 202114244.
- 130 B. Han, Y. Jin, B. Chen, W. Zhou, B. Yu, C. Wei, H. Wang, K. Wang, Y. Chen, B. Chen and J. Jiang, *Angew. Chem., Int. Ed.*, 2021, **61**, 202114244.
- 131 Y. R. Wang, H. M. Ding, X. Y. Ma, M. Liu, Y. L. Yang, Y. Chen, S. L. Li and Y. Q. Lan, *Angew. Chem., Int. Ed.*, 2022, **61**, 202114648.



- 132 S. Ali, R. Iqbal, F. Wahid, P. M. Ismail, A. Saleem, S. Ali, F. Raziq, S. Ullah, I. Ullah, Tahir, M. Zahoor, X. Wu, H. Xiao, X. Zu and L. Qiao, *Fuel Process. Technol.*, 2022, **237**, 107451.
- 133 M. Liu, Y. R. Wang, H. M. Ding, M. Lu, G. K. Gao, L. Z. Dong, Q. Li, Y. Chen, S. L. Li and Y. Q. Lan, *Sci. Bull.*, 2021, **66**, 1659–1668.
- 134 Y.-L. Yang, Y.-R. Wang, G.-K. Gao, M. Liu, C. Miao, L.-Y. Li, W. Cheng, Z.-Y. Zhao, Y. Chen, Z. Xin, S.-L. Li, D.-S. Li and Y.-Q. Lan, *Chin. Chem. Lett.*, 2022, **33**, 1439–1444.
- 135 X. F. Qiu, J. R. Huang, C. Yu, Z. H. Zhao, H. L. Zhu, Z. Ke, P. Q. Liao and X. M. Chen, *Angew. Chem., Int. Ed.*, 2022, **61**, 202206470.
- 136 L. Gong, B. Chen, Y. Gao, B. Yu, Y. Wang, B. Han, C. Lin, Y. Bian, D. Qi and J. Jiang, *Inorg. Chem. Front.*, 2022, **9**, 3217–3223.
- 137 M. Lu, M. Zhang, C. G. Liu, J. Liu, L. J. Shang, M. Wang, J. N. Chang, S. L. Li and Y. Q. Lan, *Angew. Chem., Int. Ed.*, 2021, **60**, 4864–4871.
- 138 M. Liu, X. Zhao, S. Yang, X. Yang, X. Li, J. He, G. Z. Chen, Q. Xu and G. Zeng, *ACS Appl. Mater. Interfaces*, 2023, **15**, 44384–44393.
- 139 K. S. Song, P. W. Fritz, D. F. Abbott, L. N. Poon, C. M. Caridade, F. Gandara, V. Mougel and A. Coskun, *Angew. Chem., Int. Ed.*, 2023, **62**, 202309775.
- 140 M. D. Zhang, J. R. Huang, W. Shi, P. Q. Liao and X. M. Chen, *Angew. Chem., Int. Ed.*, 2023, **62**, 202308195.
- 141 Q. J. Wu, D. H. Si, Q. Wu, Y. L. Dong, R. Cao and Y. B. Huang, *Angew. Chem., Int. Ed.*, 2023, **62**, 202215687.
- 142 Q. J. Wu, D. H. Si, S. Ye, Y. L. Dong, R. Cao and Y. B. Huang, *J. Am. Chem. Soc.*, 2023, **145**, 19856–19865.
- 143 Y. Zhang, X. Zhang, L. Jiao, Z. Meng and H. L. Jiang, *J. Am. Chem. Soc.*, 2023, **145**, 24230–24239.
- 144 C. L. Smith, R. Clowes, R. S. Sprick, A. I. Cooper and A. J. Cowan, *Sustainable Energy Fuels*, 2019, **3**, 2990–2994.
- 145 S. Wei, H. Zou, W. Rong, F. Zhang, Y. Ji and L. Duan, *Appl. Catal., B*, 2021, **284**, 119739.
- 146 K. Chen, M. Cao, G. Ni, S. Chen, H. Liao, L. Zhu, H. Li, J. Fu, J. Hu, E. Cortés and M. Liu, *Appl. Catal., B*, 2022, **306**, 121093.
- 147 Z. Wang, Y. Han, B. Li, P. Peng and S. Q. Zang, *Small*, 2023, **19**, 2301797.
- 148 R. Wang, X. Wang, W. Weng, Y. Yao, P. Kidkhunthod, C. Wang, Y. Hou and J. Guo, *Angew. Chem., Int. Ed.*, 2022, **61**, 202115503.
- 149 T. Wang, L. Xu, Z. Chen, L. Guo, Y. Zhang, R. Li and T. Peng, *Appl. Catal., B*, 2021, **291**, 120128.
- 150 Y. Song, J.-J. Zhang, Z. Zhu, X. Chen, L. Huang, J. Su, Z. Xu, T. H. Ly, C.-S. Lee, B. I. Yakobson, B. Z. Tang and R. Ye, *Appl. Catal., B*, 2021, **284**, 119750.
- 151 Qi. Li, Z.-M. Wang, Y. Chen, Y.-R. Wang, C. Guo, Q. Huang, L.-Z. Dong, S.-L. Li and Y.-Q. Lan, *J. Mater. Chem. A*, 2022, **10**, 25356–25362.
- 152 P. T. Smith, B. P. Benke, Z. Cao, Y. Kim, E. M. Nichols, K. Kim and C. J. Chang, *Angew. Chem., Int. Ed.*, 2018, **57**, 9684–9688.
- 153 X. Liu, C. Liu, X. Song, X. Ding, H. Wang, B. Yu, H. Liu, B. Han, X. Li and J. Jiang, *Chem. Sci.*, 2023, **14**, 9086–9094.
- 154 Y. Hu, S. Huang, L. J. Wayment, J. Wu, Q. Xu, T. Chang, Y.-P. Chen, X. Li, B. Andi, H. Chen, Y. Jin, H. Zhu, M. Du, S. Lu and W. Zhang, *Cell Rep. Phys. Sci.*, 2023, **4**, 101285.
- 155 N. Han, Y. Wang, L. Ma, J. Wen, J. Li, H. Zheng, K. Nie, X. Wang, F. Zhao, Y. Li, J. Fan, J. Zhong, T. Wu, D. J. Miller, J. Lu, S.-T. Lee and Y. Li, *Chem*, 2017, **3**, 652–664.

