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## A robust CdS/In<sub>2</sub>O<sub>3</sub> hierarchical heterostructure derived from a metal–organic framework for efficient visible-light photocatalytic hydrogen production†

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Building sulfide/oxide heterostructure photocatalysts is not only beneficial for the separation and transfer of photo-generated carriers in photocatalytic reactions but also an effective means of broadening the light absorption range of oxides by introducing narrow-bandgap sulfides. However, the construction of highly-efficient sulfide/oxide heterostructures with intimate and extensive contact between different components for visible-light photocatalytic hydrogen production remains a challenge. Herein, a novel sulfide/oxide heterostructure of CdS/In<sub>2</sub>O<sub>3</sub> hierarchical nanotubes was synthesized starting from a metal–organic framework (MOF). In such a heterostructure, CdS and In<sub>2</sub>O<sub>3</sub> are integrated by close contact of corresponding small nanoparticles, which is beneficial for the separation and transfer of photo-generated charge carriers. Benefiting from the intimate interface, suitable energy band structure, as well as the inherent advantages of MOFs-derived nanomaterials, the optimal CdS/In<sub>2</sub>O<sub>3</sub> hierarchical nanotubes showed the best activity in visible-light photocatalytic hydrogen production without any cocatalysts, which was 78.6 times and 16.9 times higher than activities of In<sub>2</sub>O<sub>3</sub> hierarchical nanotubes and CdS nanoparticles, respectively.

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## Introduction

In today's society, the ever-growing energy demand has become a major factor restricting the progress of human civilization. As the main energy source nowadays, fossil fuels are increasingly exhausted and new and sustainable energy resources are necessary.<sup>1–7</sup> Hydrogen production by photocatalytic water splitting is a potential strategy to convert solar energy to chemical storable and clean energy of hydrogen.<sup>8–11</sup> As a result, such a technique has attracted extensive attention to addressing the increasingly severe energy crisis. The key challenge for scientists is to develop efficient and stable photocatalysts. Up till now, various kinds of semiconductor photocatalysts, such as metal oxides (e.g. TiO<sub>2</sub><sup>12–15</sup> and SrTiO<sub>3</sub><sup>16–18</sup>), metal nitrides (e.g. Ta<sub>3</sub>N<sub>5</sub><sup>19,20</sup> and WN<sup>21</sup>), metal sulfides (e.g. Cu<sub>x</sub>S<sup>22,23</sup> and CdS<sup>24–28</sup>) and metal-free carbon nitrides,<sup>29–36</sup>

have been developed and applied in the field of solar-driven water splitting to produce hydrogen fuels. However, the photocatalytic efficiency of hydrogen production is still far lower than expected. In order to enhance the efficiency of photocatalytic reactions, various methods of catalyst preparation have been tried, including nanostructuralization,<sup>37–40</sup> doping of heteroatoms,<sup>41–44</sup> using cocatalysts,<sup>45–48</sup> building of heterostructures<sup>49–51</sup> and so on. Among them, building heterostructures is a rapid and effective method to promote photocatalytic efficiency of hydrogen production.

Building heterostructure photocatalysts is a versatile means of promoting the separation and transfer of photo-generated carriers through the interface in photocatalytic reactions.<sup>49–51</sup> On this basis, the construction of sulfide/oxide heterostructures is also in favour of broadening light absorption range of oxides by introducing narrow-bandgap sulfides, thus, obtaining highly-efficient photocatalysts with relatively high stability and low cost. Although diverse sulfide/oxide heterostructure photocatalysts have been designed and synthesized,<sup>52–55</sup> many of them do not show remarkable enhancement towards visible-light photocatalytic hydrogen production without cocatalysts. This may be primarily caused by deficient interfaces of two or more components in photocatalysts as a result of large bulk volume or unreasonable contact forms.

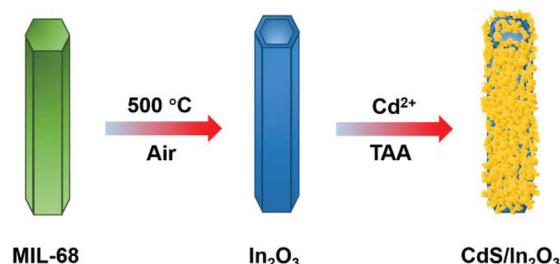
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Based on the above considerations, MOFs (metal-organic frameworks) derived heterostructures for photocatalysis have been widely studied in recent years.<sup>55–58</sup> Such heterostructures can often inherit the porous structure of original MOFs and own a lot of advantages. First, such hierarchical structures with interior voids can enhance light harvesting by extensive reflection and scattering of incident light.<sup>9</sup> Secondly, the heterostructures are composed of small nanosized subunits, offering more exposed active sites owing to large specific surface areas. Moreover, nanosized building blocks make it possible to realize intimate and extensive contact between two or more semiconductor components, which makes photo-induced carriers separate and transfer to the surface more efficiently. In recent years, MOFs-derived sulfides or sulfide/sulfide heterostructure photocatalysts for solar-to-chemical energy conversion have been widely studied.<sup>59–62</sup> In particular, the Lou's group have recently developed two In-MOF-derived hierarchical tubular heterostructures ( $\text{In}_2\text{S}_3/\text{CdIn}_2\text{S}_4$ <sup>61</sup> and  $\text{ZnIn}_2\text{S}_4/\text{In}_2\text{O}_3$ <sup>55</sup>) for efficient visible-light  $\text{CO}_2$  photoreduction. Their research provides new ideas for designing heterostructure photocatalysts. However, MOFs-derived high-efficiency sulfide/oxide heterostructures for visible-light photocatalytic hydrogen production have rarely been studied and need to be further explored.

From another perspective,  $\text{In}_2\text{O}_3$  is a common oxide semiconductor in visible-light photocatalysis with a bandgap of around 2.8 eV.<sup>39,63–65</sup> Although various  $\text{In}_2\text{O}_3$ -based heterostructures for photocatalytic hydrogen production have been synthesized, including Cr-doped  $\text{Ba}_2\text{In}_2\text{O}_5/\text{In}_2\text{O}_3$ ,<sup>66</sup>  $\text{Gd}_2\text{Ti}_2\text{O}_7/\text{In}_2\text{O}_3$ ,<sup>67</sup>  $\text{In}_2\text{O}_3/\text{La}_2\text{Ti}_2\text{O}_7$ ,<sup>68</sup>  $\text{In}_2\text{O}_3/\text{ZnO}$ ,<sup>69</sup>  $\text{In}_2\text{O}_3/\text{g-C}_3\text{N}_4$ ,<sup>70</sup>  $\text{In}_2\text{S}_3/\text{In}_2\text{O}_3$ <sup>53</sup> and  $\text{In}_2\text{S}_3/\text{CdIn}_2\text{S}_4/\text{In}_2\text{O}_3$ ,<sup>71</sup> most of them can only boost the activity of  $\text{In}_2\text{O}_3$  by no more than 30 times, even in the presence of Pt cocatalyst. On the other hand, as an efficient narrow-bandgap photocatalyst for hydrogen production, CdS suffers from low photostability and rapid recombination of photo-induced charge carriers.<sup>24–28</sup> It has been found in recent years that building heterostructures<sup>72,73</sup> or phase junctions,<sup>74</sup> homoepitaxial growth on CdS nanocrystals,<sup>75</sup> using cocatalysts (e.g. Pt<sup>76</sup> and Co(II)/Ni(II) species<sup>25</sup>), designing delicate porous nanostructures<sup>77</sup> as well as employing a redox shuttle for hole transfer<sup>78</sup> can help to solve these problems, especially low photostability of CdS. As a result, more and more 0D, 1D and 2D CdS-based heterostructures have been used for photocatalytic hydrogen production recently,<sup>79,80</sup> including graphene-CdS-MoS<sub>2</sub> composites,<sup>81</sup> CdS@MoS<sub>2</sub> core-shell nanowires<sup>82</sup> and so on. These heterostructures all showed huge promotion in photocatalytic activity and stability. In other words, CdS has become a potential candidate for the construction of efficient hydrogen production photocatalysts. Nevertheless, few attempts have been made to combine  $\text{In}_2\text{O}_3$  with CdS to construct heterostructure photocatalysts, not to mention that the interfacial effect of  $\text{In}_2\text{O}_3$  and CdS in photocatalytic hydrogen production has never been studied in the absence of cocatalysts.<sup>83</sup>

Taking all the above into account, herein, we synthesized a novel sulfide/oxide heterostructure of CdS/ $\text{In}_2\text{O}_3$  hierarchical



**Scheme 1** Schematic of the synthesis process of CdS/ $\text{In}_2\text{O}_3$  hierarchical nanotubes.

nanotubes by a facile two-step method, that is, the formation of  $\text{In}_2\text{O}_3$  hierarchical nanotubes by calcination of an In-MOF (MIL-68) followed by *in situ* growth of CdS nanoparticles *via* hydrolysis of thioacetamide (as illustrated in Scheme 1). In such a heterostructure, CdS and  $\text{In}_2\text{O}_3$  are integrated by close contact of corresponding small nanoparticles, which is beneficial to the separation and transfer of photo-generated charge carriers. In consequence, the optimal CdS/ $\text{In}_2\text{O}_3$  hierarchical nanotubes showed the best activity in visible-light photocatalytic hydrogen production without any cocatalysts, which was 78.6 times and 16.9 times greater than activities of  $\text{In}_2\text{O}_3$  hierarchical nanotubes and CdS nanoparticles, respectively.

## Experimental section

### Chemicals

$\text{In}(\text{NO}_3)_3 \cdot x\text{H}_2\text{O}$  (A.R., Aladdin), anhydrous  $\text{InCl}_3$  (A.R., Sinopharm Chemical Reagent Co. Ltd, China), urea (A.R., Xilong Chemical Co. Ltd, China), polyvinylpyrrolidone (PVP,  $M_w = 58\,000$ , A.R., Aladdin), terephthalic acid (H<sub>2</sub>BDC, A.R., Sigma-Aldrich),  $\text{CdCl}_2 \cdot 2.5\text{H}_2\text{O}$  (A.R., Sinopharm Chemical Reagent Co. Ltd, China), thioacetamide (TAA, A.R., Sinopharm Chemical Reagent Co. Ltd, China), *N,N*-dimethylformamide (DMF, A.R., Xilong Chemical Co. Ltd, China), triethanolamine (TEOA, A.R., Beijing Tongguang Fine Chemicals Company), ethanol (A.R.) and deionized water (Millipore, 18.2 MΩ cm) were used as received.

### Synthesis of $\text{In}_2\text{O}_3$ hierarchical nanotubes

The MIL-68 hexagonal prisms were first synthesized according to a reported oil bath method with some modifications.<sup>84</sup> Briefly, 468 mg of  $\text{In}(\text{NO}_3)_3 \cdot x\text{H}_2\text{O}$  and 180 mg of H<sub>2</sub>BDC were dissolved in 54 mL of DMF under continuous stirring for 15 min, followed by reacting in an oil bath (100 °C) for 1.5 h. After the resulting mixture was cooled to room temperature, the white precipitate was collected by centrifugation, washed with DMF and ethanol several times, and finally dried in vacuum at 50 °C. The  $\text{In}_2\text{O}_3$  nanotubes with a light yellow color were finally obtained by calcining the MIL-68 hexagonal prisms in air at 500 °C for 2 h with a heating rate of 5 °C min<sup>-1</sup>.

### Synthesis of CdS/In<sub>2</sub>O<sub>3</sub> hierarchical nanotubes

The *in situ* growth of CdS based on In<sub>2</sub>O<sub>3</sub> hierarchical nanotubes was achieved through the hydrolysis of TAA. In particular, 40 mg of In<sub>2</sub>O<sub>3</sub> nanotubes were dispersed in 20 mL of H<sub>2</sub>O by sonication, and 1.6 mL of CdCl<sub>2</sub> solution (1 mol L<sup>-1</sup>) was added. After stirring for 20 min, 20 mL of aqueous solution containing 240 mg TAA was added and the mixture was stirred for another 10 min. Finally, the mixture was placed into an oil bath (70 °C) and kept for 4 h. The orange-yellow product (denoted as CdS/In<sub>2</sub>O<sub>3</sub>-2 or CI-2 for short) was collected by centrifugation, washed with H<sub>2</sub>O and ethanol several times, and dried in vacuum at 40 °C. CdS/In<sub>2</sub>O<sub>3</sub>-1 (CI-1) and CdS/In<sub>2</sub>O<sub>3</sub>-3 (CI-3) were synthesized using the same procedure as that for CdS/In<sub>2</sub>O<sub>3</sub>-2, except 0.75 and 1.25 times the amount of precursors in the growth step of CdS (including CdCl<sub>2</sub> solution and TAA) were used, respectively.

### Synthesis of CdS nanoparticle aggregates

300 mg of TAA was dissolved in 50 mL of H<sub>2</sub>O, followed by the addition of 2 mL of CdCl<sub>2</sub> solution (1 mol L<sup>-1</sup>). The solution was placed into an oil bath (70 °C) and kept for 8 h. The orange product was collected by centrifugation, washed with H<sub>2</sub>O and ethanol several times, and finally dried in vacuum at 40 °C.

### Synthesis of In<sub>2</sub>O<sub>3</sub> nanorods

In<sub>2</sub>O<sub>3</sub> nanorods were synthesized according to a reported hydrothermal method.<sup>53</sup> 100 mg of InCl<sub>3</sub>, 400 mg of urea and 800 mg of PVP were dissolved in 15 mL of H<sub>2</sub>O with continuous stirring. The solution was then transferred to a 25 mL Teflon-lined stainless steel autoclave and heated at 100 °C for 24 h. After the mixture was cooled to room temperature, the white precursor was collected by centrifugation, washed with H<sub>2</sub>O and ethanol several times, and dried in vacuum at 50 °C. The In<sub>2</sub>O<sub>3</sub> nanorods with a light yellow color were finally obtained by calcining the precursor in air at 350 °C for 2 h at a heating rate of 5 °C min<sup>-1</sup>.

### Synthesis of CdS/In<sub>2</sub>O<sub>3</sub> nanorods

The synthesis of CdS/In<sub>2</sub>O<sub>3</sub> nanorods (denoted as CIR-2 for short) was the same as that of CdS/In<sub>2</sub>O<sub>3</sub>-2 (CI-2), except that In<sub>2</sub>O<sub>3</sub> nanorods rather than hierarchical nanotubes were used.

### Characterization

Transmission electron microscopy (TEM), high-resolution transmission electron microscopy (HRTEM), high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) and energy-dispersive X-ray spectroscopy (EDS) were carried out using FEG-TEM (JEM-2100F, JEOL, Japan) operated at 200 kV. Scanning electron microscopy (SEM) was carried out using a FE-SEM (Hitachi S-4800, Japan). X-ray diffraction (XRD) patterns were obtained on a D/MAX-2000 diffractometer (Rigaku, Japan) using Cu-K $\alpha$  radiation. Inductively coupled plasma-

atomic emission spectroscopy (ICP-AES) analysis was performed on a Profile Spec ICP-AES spectrometer (Leeman, USA). X-ray photoelectron spectroscopy (XPS) analysis was performed on an Axis Ultra XPS spectrometer (Kratos, U.K.). The C 1s line at 284.8 eV was used to calibrate the binding energies. Nitrogen adsorption–desorption isotherm experiments were performed at 77 K on an Accelerated Surface Area & Porosimetry system (ASAP 2010, Micromeritics, USA) to determine the Brunauer–Emmett–Teller (BET) surface area and pore size distribution. UV-vis diffuse reflectance spectra (UV-vis DRS) were acquired using a UV-3600 Plus UV-vis-NIR spectrometer (Shimadzu, Japan) equipped with an integrating sphere, and BaSO<sub>4</sub> was used as a reference. Photoluminescence (PL) spectra were obtained using a FLS980 steady-state and time-resolved fluorescence spectrometer (Edinburgh Instruments, U.K.) at room temperature. The excitation wavelength was 300 nm for steady-state PL. Time-resolved PL decay spectra were recorded at 545 nm under the excitation of a 375 nm laser source.

Electrochemical measurements were performed in a standard three-electrode system. The electrolyte was Na<sub>2</sub>SO<sub>4</sub> solution (0.5 M, pH ~ 6.1). A Pt plate and a KCl-saturated Ag/AgCl electrode were used as the counter and the reference electrode, respectively. The working electrode was prepared as described below. First, the as-prepared photocatalyst was dispersed in a mixed solvent of H<sub>2</sub>O, ethanol and Nafion (8 : 2 : 1 by volume) to form a suspension with a concentration of *ca.* 9 mg mL<sup>-1</sup>. Then 35  $\mu$ L of the suspension was dropped onto a fixed area (*ca.* 0.2826 cm<sup>2</sup>) of an ITO glass. Finally, the ITO glass was naturally dried to obtain the working electrode. Transient photocurrents were measured at 0.6 V (vs. Ag/AgCl) using a 300 W Xe lamp (with an intensity of *ca.* 100 mW cm<sup>-2</sup>) as a light source. Electrochemical impedance spectroscopy (EIS) measurements were performed at a potential of 0.3 V (vs. Ag/AgCl) and an amplitude of 10 mV. The frequency range was from 100 kHz to 10 mHz. Mott–Schottky tests were conducted with different frequencies of 0.5, 1 and 2 kHz in the dark.

### Photocatalytic hydrogen production from water

A quartz photoreactor attached to a closed photocatalytic system equipped with an online gas chromatograph (GC-7806) was used to evaluate the activity of photocatalytic hydrogen production. 20 mg of the catalyst was suspended in a mixture of 72 mL of H<sub>2</sub>O and 8 mL of TEOA. The temperature of the reactor was kept at 288 K by circulating cooling water. Before light irradiation, the system was evacuated several times to remove the air inside. Then the reactor was illuminated by a 300 W Xe lamp (Microsolar 300, PerfectLight) equipped with a UV cut-off filter ( $\lambda$  > 400 nm). The light intensity was approximately 500 mW cm<sup>-2</sup>. Hydrogen was detected by a thermal conductivity detector (TCD) and Ar was used as the carrier gas.

The apparent quantum efficiency (AQE) of photocatalysts was measured using a monochromatic irradiation light of

420 nm by a 300 W Xe lamp equipped with a band-pass filter. The AQE was calculated as follows:

$$\begin{aligned} \text{AQE} &= \frac{\text{number of reacted electrons}}{\text{number of incident photons}} \times 100\% \\ &= \frac{\text{number of generated hydrogen} \times 2}{\text{number of incident photons}} \times 100\% \\ &= \frac{n_{\text{hydrogen}} \times N_A \times 2}{PSt\lambda/hc} \times 100\%, \end{aligned}$$

wherein,  $n_{\text{hydrogen}}$  is the mole number of generated hydrogen,  $N_A$  is the Avogadro's constant ( $6.02 \times 10^{23} \text{ mol}^{-1}$ ),  $P$  is the power of Xe lamp at 420 nm (*ca.* 60 mW cm $^{-2}$ ),  $S$  is the illumination area ( $\sim 9 \text{ cm}^2$ ),  $t$  is the photocatalytic reaction time (3600 s),  $\lambda$  is the wavelength ( $420 \times 10^{-9} \text{ m}$ ),  $h$  is the Planck constant ( $6.626 \times 10^{-34} \text{ J s}^{-1}$ ), and  $c$  is the speed of light ( $3 \times 10^8 \text{ m s}^{-1}$ ).

## Results and discussion

### Synthesis, characterization and photocatalytic activity towards hydrogen production

In the first synthesis step,  $\text{In}_2\text{O}_3$  hierarchical nanotubes were formed after the calcination of the MIL-68 hexagonal prisms. For  $\text{In}_2\text{O}_3$  hierarchical nanotubes and CdS nanoparticle aggregates, all the main diffraction peaks in the XRD patterns (Fig. S1a and S1b $\dagger$ ) are indexed to cubic  $\text{In}_2\text{O}_3$  (JCPDS no. 01-089-4595) and hexagonal CdS (JCPDS no. 00-041-1049). After the growth of CdS, XRD patterns of the as-prepared CdS/ $\text{In}_2\text{O}_3$  heterostructures are presented in Fig. 1. For all heterostructures, the diffraction peaks reveal the features of both cubic  $\text{In}_2\text{O}_3$  and hexagonal CdS without obvious impurities such as  $\text{In}_2\text{S}_3$ . Moreover, the peak intensity ratio of CdS to  $\text{In}_2\text{O}_3$  in the heterostructures increases in the order of CdS/ $\text{In}_2\text{O}_3$ -1, CdS/ $\text{In}_2\text{O}_3$ -2 and CdS/ $\text{In}_2\text{O}_3$ -3, which is in accordance with the variation trend of weight ratio of CdS and  $\text{In}_2\text{O}_3$  obtained by ICP-AES analysis (*ca.* 0.375, 0.709 and 0.952 for CdS/ $\text{In}_2\text{O}_3$ -1, CdS/ $\text{In}_2\text{O}_3$ -2 and CdS/ $\text{In}_2\text{O}_3$ -3).

SEM images of the MIL-68 hexagonal prisms are shown in Fig. S2. $\dagger$  From the SEM image shown in Fig. 2a, the as-prepared  $\text{In}_2\text{O}_3$  by the calcination of MIL-68 are nanotubes in shape, with a width of 1 micrometer around and a length of several micrometers for most nanotubes. By close observation

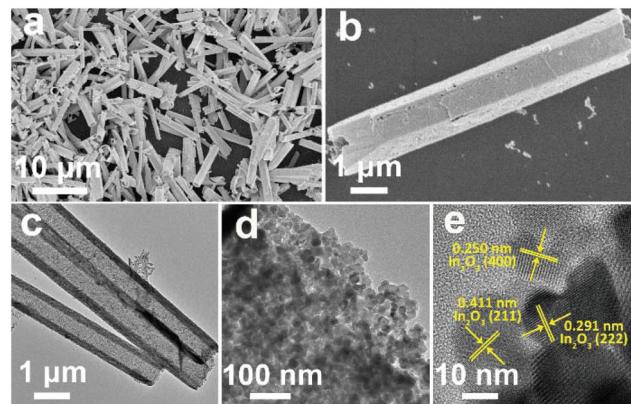


Fig. 2 (a, b) SEM, (c, d) TEM and (e) HRTEM images of  $\text{In}_2\text{O}_3$  nanotubes.

of the magnified SEM image (Fig. 2b), it can be found that the surface of the  $\text{In}_2\text{O}_3$  nanotubes is not integrated. On the contrary, the  $\text{In}_2\text{O}_3$  nanotubes are made up of small nanoparticles, as a result of which a slight unevenness of the surface is observed. The TEM image in Fig. 2c confirms such tubular morphology. Compared with the interior of the  $\text{In}_2\text{O}_3$  nanotubes, higher contrast for the edge reflects the hollow feature of nanotubes. As illustrated by magnified TEM image of the nanotube mouth (Fig. 2d), the constituent nanoparticles are approximately two dozen nanometers with irregular shapes. Furthermore, the distinct lattice fringes from the HRTEM image (Fig. 2e) show that the as-prepared  $\text{In}_2\text{O}_3$  nanotubes are well crystallized after heat treatment of the In-MOF. The interplanar distances of about 0.411 nm, 0.291 nm and 0.250 nm correspond to the (211), (222) and (400) crystal planes of cubic  $\text{In}_2\text{O}_3$ , respectively.

As the surface of the  $\text{In}_2\text{O}_3$  nanotubes is negatively charged, proven by negative zeta potential (Fig. S3 $\dagger$ ),  $\text{Cd}^{2+}$  can be adsorbed by the  $\text{In}_2\text{O}_3$  nanotubes through electrostatic interactions<sup>85</sup> and then react with a sulfur source to form CdS. After the growth of CdS, the as-obtained CdS/ $\text{In}_2\text{O}_3$  nanotubes experience noticeable changes in morphology. As shown in Fig. 3a and b, the inner and outer surfaces of CdS/ $\text{In}_2\text{O}_3$ -2 are quite rough compared with  $\text{In}_2\text{O}_3$  nanotubes, although the tubular shape remains unchanged. CdS/ $\text{In}_2\text{O}_3$ -1 and CdS/ $\text{In}_2\text{O}_3$ -3 show similar morphology, except with different degrees of roughness (Fig. S4 $\dagger$ ). The TEM image in Fig. 3c further validates such a change. The edges of  $\text{In}_2\text{O}_3$  nanotubes are obviously uneven with nanoparticles different from  $\text{In}_2\text{O}_3$  nanoparticles in shape (shown in Fig. 3d). In addition, the arrangement of nanoparticles in the structure of nanotubes becomes more compact, indicating that the as-grown CdS nanoparticles are in close proximity to  $\text{In}_2\text{O}_3$  nanoparticles in the CdS/ $\text{In}_2\text{O}_3$  nanotubes. SEM images of CdS/ $\text{In}_2\text{O}_3$ -2 at different time in the growth process of CdS are shown in Fig. S5. $\dagger$  It is evident that the nanotube morphology of  $\text{In}_2\text{O}_3$  is unchanged during the formation process of CdS nanoparticles. While pure CdS is composed of agglomerates as large as several micrometers, which, in turn, are composed of

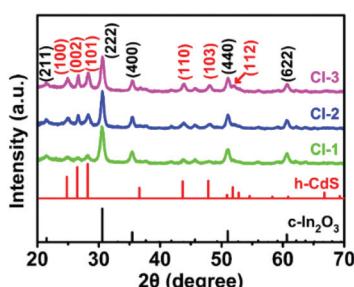


Fig. 1 XRD patterns of the as-prepared CdS/ $\text{In}_2\text{O}_3$  heterostructures.

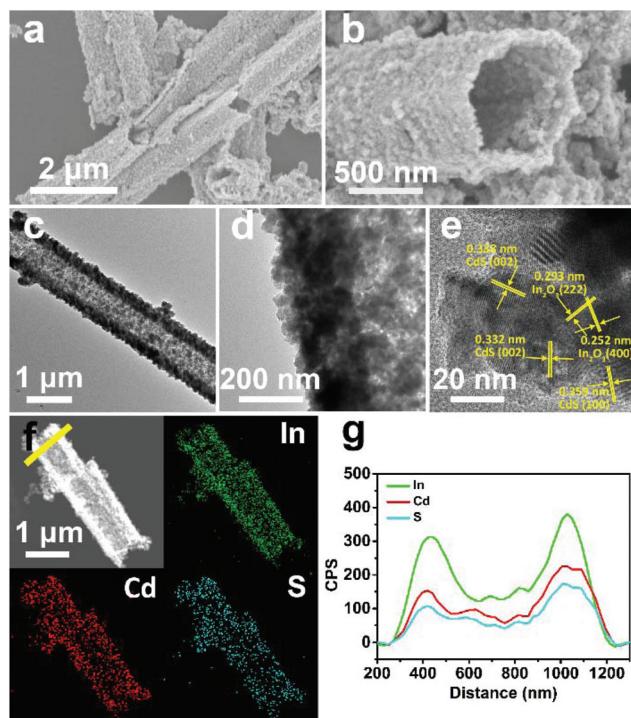


Fig. 3 (a, b) SEM, (c, d) TEM, (e) HRTEM and (f) HAADF-STEM and EDS mapping images of CdS/In<sub>2</sub>O<sub>3</sub>-2; (g) EDS line scan profiles of CdS/In<sub>2</sub>O<sub>3</sub>-2 obtained along the yellow line in (f).

nanoparticles (Fig. S6†), the CdS nanoparticles in CdS/In<sub>2</sub>O<sub>3</sub> nanotubes are apparently more dispersive. Such a phenomenon reveals that In<sub>2</sub>O<sub>3</sub> nanotubes are ideal supports for the decentralized growth of CdS nanoparticles without the assistance of surfactants. This may be beneficial to the enhancement of surface area and the exposure of more active sites compared with pure CdS. To verify the combination of CdS and In<sub>2</sub>O<sub>3</sub>, HRTEM characterization was carried out (Fig. 3e). The lattice fringes corresponding to (100) and (002) crystal planes of hexagonal CdS can be found around In<sub>2</sub>O<sub>3</sub> nanoparticles, reflecting the formation of CdS/In<sub>2</sub>O<sub>3</sub> heterostructure to some extent. In addition, two sets of diffraction fringes, belonging to In<sub>2</sub>O<sub>3</sub> and CdS in selected area electron diffraction (SAED) pattern (Fig. S7†), support the formation of CdS/In<sub>2</sub>O<sub>3</sub> heterostructure. Elemental mapping was conducted to further show that In, Cd and S are evenly distributed in the heterostructure (Fig. 3f). Moreover, the line scan result in Fig. 3g not only verifies the hollow feature of CdS/In<sub>2</sub>O<sub>3</sub> nanotubes again, but also shows uniform variation tendency of the element In, Cd and S. Elemental mapping and line scan both lead to the fact that homogeneous CdS/In<sub>2</sub>O<sub>3</sub> heterostructure can be obtained by *in situ* growth of CdS nanoparticles on the basis of In<sub>2</sub>O<sub>3</sub> nanotubes. In such a hierarchical heterostructure, CdS and In<sub>2</sub>O<sub>3</sub> are integrated by close contact of corresponding small nanoparticles, which ensures large interfaces of these two semiconductor components. Compared to other CdS/In<sub>2</sub>O<sub>3</sub> photocatalysts, no templates<sup>86</sup> or substrates<sup>87</sup> were used in our synthetic procedure. In addition, *in situ*

growth of CdS means that we did not need to pre-synthesize CdS nanoparticles and combine these two semiconductor nanostructures using an extra step.<sup>83</sup> As a whole, our synthetic procedure was simplified to a great extent.

X-ray photoelectron spectroscopy (XPS) analysis was performed to determine the chemical composition and valence states of the elements in CdS/In<sub>2</sub>O<sub>3</sub>-2. The survey-scan spectrum of CdS/In<sub>2</sub>O<sub>3</sub>-2 (Fig. S8†) reveals the existence of In, O, Cd, S and C on the surface and sub-surface of CdS/In<sub>2</sub>O<sub>3</sub> nanotubes. For In 3d core-level spectrum (Fig. 4a), two peaks centered at 452.1 and 444.5 eV correspond to In 3d<sub>3/2</sub> and In 3d<sub>5/2</sub> peaks of In<sup>3+</sup>, respectively.<sup>55,66</sup> It is noteworthy that the peak positions of In 3d<sub>3/2</sub> and In 3d<sub>5/2</sub> for pure In<sub>2</sub>O<sub>3</sub> are 451.6 eV and 444.0 eV, respectively. The binding energy shift of *ca.* 0.5 eV means that the chemical environment of Indium in In<sub>2</sub>O<sub>3</sub> nanotubes has been changed after the formation of CdS/In<sub>2</sub>O<sub>3</sub> heterostructure, reflecting the strong interaction between CdS and In<sub>2</sub>O<sub>3</sub> in their interfacial zone.<sup>71,88</sup> For the O 1s core-level spectrum (Fig. 4b), the peak positioned at 530.0 eV is attributed to the lattice oxygen, while the peak positioned at 531.8 eV is attributed to the surface hydroxyl oxygen.<sup>88,89</sup> The Cd 3d high resolution XPS spectrum (Fig. 4c) shows two peaks at 412.4 and 405.6 eV, which are assigned to Cd 3d<sub>3/2</sub> and Cd 3d<sub>5/2</sub> peaks of Cd<sup>2+</sup>, respectively.<sup>90,91</sup> As for S 2p spectrum (Fig. 4d), two peaks at 163.2 and 162.0 eV are respectively ascribed to S 2p<sub>1/2</sub> and S 2p<sub>3/2</sub> peaks of S<sup>2-</sup>.<sup>91</sup> The binding energies of Cd and S are in good agreement with a previous report about CdS nanoparticles,<sup>91</sup> indicating the successive introduction of CdS in our heterostructure. All the above XPS results further confirm the formation of CdS/In<sub>2</sub>O<sub>3</sub> heterostructure.

UV-vis diffuse reflectance spectra (UV-vis DRS) can reflect the optical absorption property of semiconductors. Fig. 5a depicts the DRS of In<sub>2</sub>O<sub>3</sub>, CdS and CdS/In<sub>2</sub>O<sub>3</sub> heterostructures.

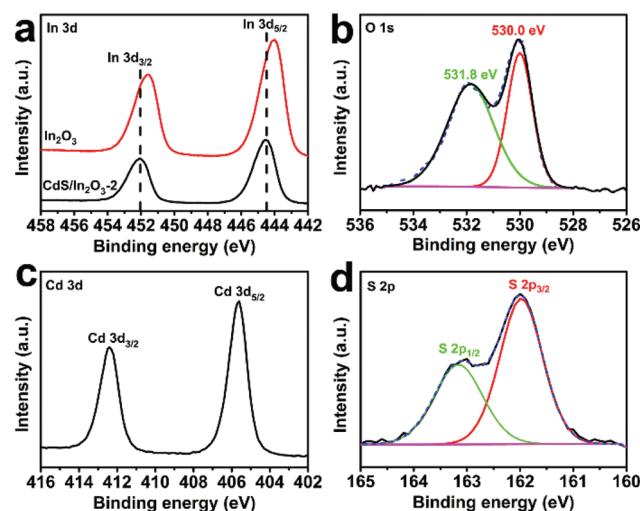


Fig. 4 High resolution XPS spectra of (a) In 3d, (b) O 1s, (c) Cd 3d and (d) S 2p in CdS/In<sub>2</sub>O<sub>3</sub>-2. The red curve in (a) is the high resolution XPS spectrum of In 3d in In<sub>2</sub>O<sub>3</sub> nanotubes.

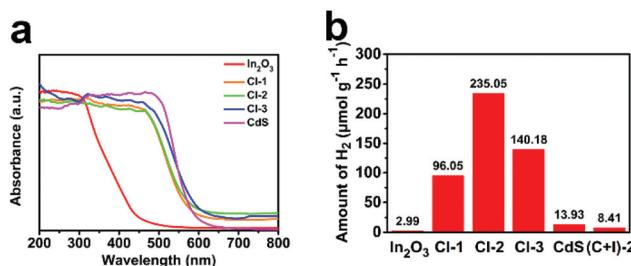


Fig. 5 (a) UV-vis DRS of  $\text{In}_2\text{O}_3$ ,  $\text{CdS}$  and  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructures; (b) photocatalytic activities of different samples towards visible-light hydrogen production from water.

Moreover, band gap energy ( $E_g$ ) of the samples can be estimated utilizing Tauc plots:  $(\alpha h\nu)^{1/r} = C(h\nu - E_g)$ , where  $\alpha$ ,  $\nu$  and  $C$  are the absorption coefficient, light frequency and a constant, respectively;  $r$  is 1/2 (for direct band gap semiconductors) or 2 (for indirect band gap semiconductors). The results are shown in Fig. S9.† Pure  $\text{In}_2\text{O}_3$  nanotubes mainly absorb UV light ( $\lambda < 400$  nm) and are nearly incapable of absorbing visible light with a wavelength greater than 500 nm, which is unfavourable for visible-light photocatalysis.  $E_g$  of  $\text{In}_2\text{O}_3$  nanotubes is estimated to be 2.75 eV. In contrast, pure  $\text{CdS}$  has a wider absorption range in the visible region and its  $E_g$  is only 2.25 eV. For all  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructures, the light absorption capacity is very close to that of  $\text{CdS}$  ( $E_g = 2.30$  eV, 2.28 eV and 2.23 eV for  $\text{CdS}/\text{In}_2\text{O}_3$ -1,  $\text{CdS}/\text{In}_2\text{O}_3$ -2 and  $\text{CdS}/\text{In}_2\text{O}_3$ -3, respectively), which is undoubtedly an advantage compared with pure  $\text{In}_2\text{O}_3$  nanotubes.

The activities of different samples for visible-light photocatalytic hydrogen production are shown in Fig. 5b. All the photocatalytic tests were performed without the assistance of cocatalysts. For pure  $\text{In}_2\text{O}_3$  nanotubes and  $\text{CdS}$  nanoparticle aggregates, the activities were fairly low, which is mainly attributed to fast recombination of photo-generated electron–hole pairs in single-component semiconductor photocatalysts. In addition, weak light absorption ability of  $\text{In}_2\text{O}_3$  nanotubes (shown in Fig. 5a) and obvious aggregation of  $\text{CdS}$  nanoparticles (shown in Fig. S6†) become their respective constraints. Intriguingly, for all  $\text{CdS}/\text{In}_2\text{O}_3$  hierarchical nanotubes, significant photocatalytic activity improvements were observed in comparison to pure  $\text{In}_2\text{O}_3$  or  $\text{CdS}$ . Among them,  $\text{CdS}/\text{In}_2\text{O}_3$ -2 exhibited the most superior activity (235.05  $\mu\text{mol g}^{-1} \text{h}^{-1}$ ), which was approximately 78.6 times and 16.9 times greater than that of  $\text{In}_2\text{O}_3$  and  $\text{CdS}$ , respectively. The wavelength-dependent hydrogen generation activity of  $\text{CdS}/\text{In}_2\text{O}_3$ -2 was also evaluated by using different cut-off filters ( $\lambda > 400$  nm, 420 nm, 470 nm, 510 nm and 580 nm). The trend of hydrogen evolution matched well with photoabsorption of  $\text{CdS}/\text{In}_2\text{O}_3$ -2 (Fig. S10†), revealing that the hydrogen evolution reaction was indeed driven by the absorbed incident photons of the photocatalyst. The apparent quantum efficiency (AQE) of  $\text{CdS}/\text{In}_2\text{O}_3$ -2 at 420 nm was measured to be 0.2%, comparable with some other  $\text{In}_2\text{O}_3$ -based photocatalysts using Pt cocatalyst.<sup>66,68</sup> The comparison of activity between  $\text{CdS}/\text{In}_2\text{O}_3$ -2 and other  $\text{In}_2\text{O}_3$ -

based photocatalysts are listed in Table S1.† It is noticeable in Fig. 5b that when physical mixture of  $\text{CdS}$  and  $\text{In}_2\text{O}_3$  having the same ratio of  $\text{CdS}$  and  $\text{In}_2\text{O}_3$  with  $\text{CdS}/\text{In}_2\text{O}_3$ -2 (denoted as (C + I)-2) was used, the activity was still very low, leading to the fact that the formed well-defined interface between  $\text{CdS}$  and  $\text{In}_2\text{O}_3$  is responsible for the huge activity improvement. For comparison, the *in situ* growth of  $\text{CdS}$  was also conducted under the same conditions as that of  $\text{CdS}/\text{In}_2\text{O}_3$ -2, using solid  $\text{In}_2\text{O}_3$  nanorods (Fig. S11†) as supports. Although  $\text{CdS}$  nanoparticles were also successfully grown on the surface of  $\text{In}_2\text{O}_3$  nanorods, the size of them was obviously larger and more uneven than those of  $\text{CdS}/\text{In}_2\text{O}_3$ -2 (see Fig. S12, S13† and Fig. 3). In consequence, the contact of  $\text{CdS}$  and  $\text{In}_2\text{O}_3$  was insufficient. As expected, the activity of  $\text{CdS}/\text{In}_2\text{O}_3$  nanorods was far lower than the activity of  $\text{CdS}/\text{In}_2\text{O}_3$ -2 (Fig. S14†). This further revealed the advantages of our hollow and hierarchical heterostructure from the viewpoint of both synthetic chemistry and catalytic properties.

#### Explanations for the elevated activity of $\text{CdS}/\text{In}_2\text{O}_3$ heterostructure

In order to explain the reasons for the good activity of  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructures, nitrogen adsorption–desorption isotherm measurements were first conducted to obtain the BET surface area of the  $\text{CdS}/\text{In}_2\text{O}_3$  samples. As shown in Fig. 6a, the BET surface area of  $\text{CdS}/\text{In}_2\text{O}_3$ -2 is  $66.58 \text{ m}^2 \text{ g}^{-1}$ , larger than that of  $\text{In}_2\text{O}_3$  ( $47.53 \text{ m}^2 \text{ g}^{-1}$ ) and  $\text{CdS}$  ( $12.18 \text{ m}^2 \text{ g}^{-1}$ ) (see Fig. S15†). In other words,  $\text{CdS}/\text{In}_2\text{O}_3$ -2 is likely to expose more catalytic active sites. In addition, the BET surface areas of  $\text{CdS}/\text{In}_2\text{O}_3$ -1 and  $\text{CdS}/\text{In}_2\text{O}_3$ -3 are  $52.92 \text{ m}^2 \text{ g}^{-1}$  and  $44.91 \text{ m}^2 \text{ g}^{-1}$ , respectively (shown in Fig. S15†). These results show that when the ratio of  $\text{CdS}$  in the heterostructure increases, the BET surface area can be increased at the primary stage. However, excessive loading of  $\text{CdS}$  ( $\text{CdS}/\text{In}_2\text{O}_3$ -3, for example)

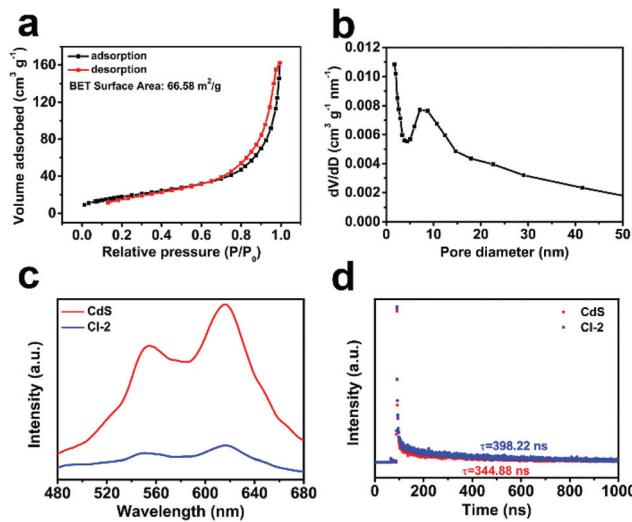


Fig. 6 (a) Nitrogen adsorption–desorption isotherm and (b) pore size distribution of  $\text{CdS}/\text{In}_2\text{O}_3$ -2; (c) steady-state PL spectra and (d) PL decay spectra of  $\text{CdS}$  and  $\text{CdS}/\text{In}_2\text{O}_3$ -2.

will block the surface of  $\text{In}_2\text{O}_3$  hierarchical nanotubes, resulting in the decrease of surface area. This is in accordance with the inferior activity of  $\text{CdS}/\text{In}_2\text{O}_3\text{-}3$  compared with activity of  $\text{CdS}/\text{In}_2\text{O}_3\text{-}2$ , as shown in Fig. 5b. Besides, the pore size distribution curve (Fig. 6b) indicates the porous nature of  $\text{CdS}/\text{In}_2\text{O}_3\text{-}2$  with mesopores around 10 nm in size ( $\text{CdS}/\text{In}_2\text{O}_3\text{-}1$  and  $\text{CdS}/\text{In}_2\text{O}_3\text{-}3$  also have similar mesoporous structure, see Fig. S15†), which is characteristic of MOFs-derived nanostructures<sup>60,61</sup> and consistent with the electron microscope data. Although  $\text{CdS}$  nanoparticle agglomerates also contain mesopores of around 10 nm in size (Fig. S15†), the small surface area as well as the fast charge recombination for single-component photocatalysts restrain their activity. In all, such porous structure of  $\text{CdS}/\text{In}_2\text{O}_3\text{-}2$  with a fairly large surface area is beneficial to the utilization of incident light, as well as the transfer and adsorption of reactive species in the photocatalytic process. Nevertheless, the increase of surface area is limited in  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructures compared with  $\text{In}_2\text{O}_3$  nanotubes if we take the huge enhancement of photocatalytic activity into consideration. In other words, there must be other primary reasons for the elevated activity of  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructure.

To analyze the reasons from the perspective of charge carrier separation, photoluminescence (PL) spectra of different photocatalysts were obtained. Compared with pure  $\text{CdS}$ , the weaker steady-state fluorescence intensity for  $\text{CdS}/\text{In}_2\text{O}_3\text{-}2$  (shown in Fig. 6c) reveals that photo-generated electron–hole pairs are recombined with greater difficulty in the form of light radiation, indicating efficient charge separation in the heterostructures under excitation light.<sup>72,92</sup> Moreover, the average PL lifetime calculated from PL decay spectra (Fig. 6d) is 398.22 ns for  $\text{CdS}/\text{In}_2\text{O}_3\text{-}2$ , longer than that of  $\text{CdS}$  (344.88 ns). Longer PL lifetimes mean slower kinetics of charge carrier recombination, which is undoubtedly in favour of the photocatalytic utilization of photo-generated carriers.<sup>40</sup>

Transient photocurrent and electrochemical impedance spectroscopy (EIS) measurements were carried out to provide more evidence for the enhanced photocatalytic activities of the  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructures. As illustrated in Fig. 7a, the photocurrent of  $\text{CdS}$  or  $\text{In}_2\text{O}_3$  alone under the irradiation of a 300 W Xe lamp is rather small, while the photocurrent is dramatically increased for all  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructures. Such a difference indicates that in  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructures, the separation and transfer of the photo-induced carriers are much faster, which is in favour of the photocatalytic process without doubt.<sup>92</sup> In addition, smaller semicircle diameters of EIS Nyquist plots for  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructures (Fig. 7b) suggest that  $\text{CdS}/\text{In}_2\text{O}_3$  nanotubes with a lower charge transfer resistance are more beneficial to the transfer of charge carriers in photocatalysis.<sup>55,85</sup>  $\text{CdS}/\text{In}_2\text{O}_3\text{-}2$ , which shows the highest photocurrent and the smallest semicircle diameter of EIS Nyquist plot, thus possesses the best photocatalytic activity, as shown in Fig. 5b.

Mott-Schottky tests were further conducted to verify semiconductor types and obtain the flat band potential of  $\text{In}_2\text{O}_3$  and  $\text{CdS}$ . In Mott-Schottky plots (Fig. 7c and d), the slope of

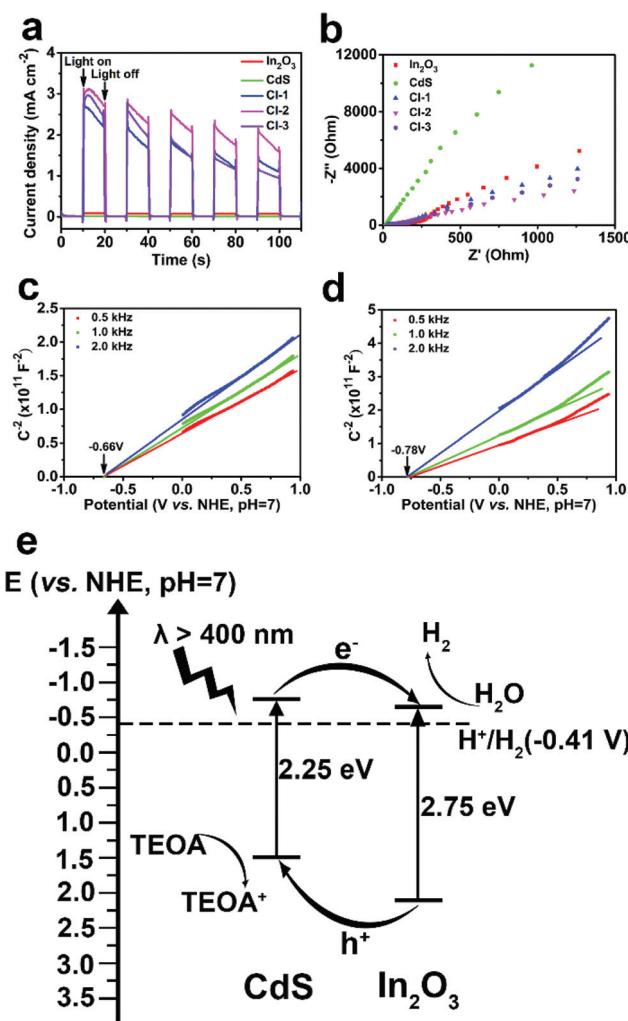
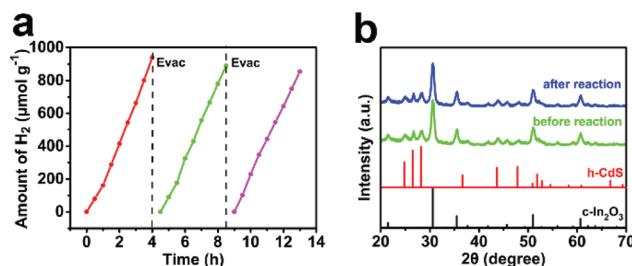


Fig. 7 (a) Transient photocurrent responses and (b) EIS Nyquist plots of  $\text{In}_2\text{O}_3$ ,  $\text{CdS}$  and  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructures; Mott-Schottky plots of (c)  $\text{In}_2\text{O}_3$  nanotubes and (d)  $\text{CdS}$  nanoparticles; (e) schematic diagram of the band structure of  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructure.

the extension line of the straight-line portion is positive for  $\text{In}_2\text{O}_3$  and  $\text{CdS}$ , revealing that both are n-type semiconductors.<sup>85,93</sup> Furthermore, the flat band potentials are  $-0.66$  and  $-0.78\text{ V}$  (vs. NHE, pH = 7) for  $\text{In}_2\text{O}_3$  and  $\text{CdS}$ , respectively. It is reasonable to use the flat band potential of an n-type semiconductor to evaluate the reducing capacity of the electrons at the conduction band minimum (CBM).<sup>94,95</sup> Both potentials are more negative than the reduction potential of  $\text{H}_2\text{O}$  ( $-0.41\text{ V}$  vs. NHE, pH = 7), indicating that both  $\text{In}_2\text{O}_3$  and  $\text{CdS}$  are potential catalysts for hydrogen production from  $\text{H}_2\text{O}$ . Combined with  $E_g$  obtained from UV-vis DRS, the schematic diagram of the band structure of  $\text{CdS}/\text{In}_2\text{O}_3$  heterostructure can be drawn (Fig. 7e). It is evident that  $\text{CdS}/\text{In}_2\text{O}_3$  is a type-II heterostructure.<sup>49</sup> Under visible light,  $\text{In}_2\text{O}_3$  and  $\text{CdS}$  can generate electron–hole pairs which are then very likely to be separated efficiently through the staggered band structure. In other words, the photo-induced electrons are collected at



**Fig. 8** (a) Photocatalytic stability of CdS/In<sub>2</sub>O<sub>3</sub>-2 towards visible-light hydrogen production; (b) XRD patterns of CdS/In<sub>2</sub>O<sub>3</sub>-2 before and after the photocatalytic cyclic test.

the bottom of conduction band of In<sub>2</sub>O<sub>3</sub>, while the photo-induced holes are collected at the top of valence band of CdS. Intimate and extensive contact of In<sub>2</sub>O<sub>3</sub> and CdS nanoparticles, described above, enlarges such an effect in a heterostructure. As a consequence, the recombination of photo-generated carriers is suppressed to a great extent, leading to the promotion of photocatalytic efficiency as more photo-induced charge carriers can transfer to the surface of photocatalysts to take part in the electrocatalytic process.

In summary, from the results of PL spectroscopy, transient photocurrent, electrochemical impedance spectroscopy as well as band structure analysis, we could deduce that the formation of a type-II heterostructure is the primary reason for the elevated photocatalytic activity of CdS/In<sub>2</sub>O<sub>3</sub> heterostructures. What is more, the hollow and hierarchical tubular morphology composed of small In<sub>2</sub>O<sub>3</sub> and CdS nanoparticles with intimate and extensive contact enlarges such an effect of heterostructure. As a result, separation and transfer of photo-generated carriers is tremendously improved in the heterostructures, leading to the huge enhancement of photocatalytic activity compared with pure In<sub>2</sub>O<sub>3</sub> or CdS.

### Photocatalytic stability test

Finally, the photocatalytic stability of CdS/In<sub>2</sub>O<sub>3</sub>-2 was also explored. It was verified that the activity of CdS/In<sub>2</sub>O<sub>3</sub>-2 remained relatively stable after being recycled 3 times (each one was as long as 4 h, seen in Fig. 8a). Besides, XRD and SEM analysis of CdS/In<sub>2</sub>O<sub>3</sub>-2 after the photocatalytic cyclic test did not show obvious changes (Fig. 8b and Fig. S16†), supporting the superior stability of such a heterostructure photocatalyst.

## Conclusions

In summary, robust CdS/In<sub>2</sub>O<sub>3</sub> hierarchical nanotubes were synthesized by a facile two-step method, that is, the formation of In<sub>2</sub>O<sub>3</sub> hierarchical nanotubes by calcination of an In-MOF followed by *in situ* growth of CdS nanoparticles by a low-temperature oil bath method. The as-prepared CdS/In<sub>2</sub>O<sub>3</sub> heterostructure is made up of small In<sub>2</sub>O<sub>3</sub> and CdS nanoparticles with intimate and extensive contacts. Such a heterostructure not only ensures large interfaces between these two semi-

conductor components which are beneficial to the separation and transfer of photo-induced charge carriers through forming a type-II heterostructure, but also possesses advantages of hierarchical nanostructures, *i.e.*, enhancing light absorption and increasing the number of catalytically-active sites. Therefore, the optimal CdS/In<sub>2</sub>O<sub>3</sub>-2 hierarchical nanotubes exhibited the best activity in visible-light photocatalytic hydrogen production without any cocatalysts, which was 78.6 times and 16.9 times greater than activities of In<sub>2</sub>O<sub>3</sub> hierarchical nanotubes and CdS nanoparticles, respectively. This work not only explores the interfacial effect of In<sub>2</sub>O<sub>3</sub> and CdS on visible-light photocatalytic hydrogen production without cocatalysts, but also provides a new idea for synthesizing highly-efficient sulfide/oxide heterostructure photocatalysts starting from MOFs, and for maximizing the interfacial effect of heterostructure photocatalysts on photocatalytic hydrogen evolution.

## Conflicts of interest

There are no conflicts to declare.

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## Notes and references

- 1 P. L. Liao and E. A. Carter, *Chem. Soc. Rev.*, 2013, **42**, 2401–2422.
- 2 W. G. Tu, Y. Zhou and Z. G. Zou, *Adv. Mater.*, 2014, **26**, 4607–4626.
- 3 D. Chen, X. G. Zhang and A. F. Lee, *J. Mater. Chem. A*, 2015, **3**, 14487–14516.
- 4 S. J. Xie, Q. H. Zhang, G. D. Liu and Y. Wang, *Chem. Commun.*, 2016, **52**, 35–59.
- 5 X. X. Chang, T. Wang and J. L. Gong, *Energy Environ. Sci.*, 2016, **9**, 2177–2196.
- 6 G. X. Zhao, X. B. Huang, X. X. Wang and X. K. Wang, *J. Mater. Chem. A*, 2017, **5**, 21625–21649.
- 7 J. Di, J. Xiong, H. M. Li and Z. Liu, *Adv. Mater.*, 2018, **30**, 1704548.
- 8 N. Q. Wu, *Nanoscale*, 2018, **10**, 2679–2696.
- 9 X. Li, J. G. Yu and M. Jaroniec, *Chem. Soc. Rev.*, 2016, **45**, 2603–2636.
- 10 J. W. Fu, J. G. Yu, C. J. Jiang and B. Cheng, *Adv. Energy Mater.*, 2018, **8**, 1701503.

11 S. S. Chen, T. Takata and K. Domen, *Nat. Rev. Mater.*, 2017, **2**, 17050.

12 W. Y. Hu, W. Zhou, K. F. Zhang, X. C. Zhang, L. Wang, B. J. Jiang, G. H. Tian, D. Y. Zhao and H. G. Fu, *J. Mater. Chem. A*, 2016, **4**, 7495–7502.

13 X. N. Ren, Z. Y. Hu, J. Jin, L. Wu, C. Wang, J. Liu, F. Liu, M. Wu, Y. Li, G. Van Tendeloo and B. L. Su, *ACS Appl. Mater. Interfaces*, 2017, **9**, 29687–29698.

14 X. L. Weng, Y. L. Zhang, F. Bi, F. Dong, Z. B. Wu and J. A. Darr, *J. Mater. Chem. A*, 2017, **5**, 23766–23775.

15 J. K. Zhang, Z. B. Yu, Z. Gao, H. B. Ge, S. C. Zhao, C. Q. Chen, S. Chen, X. L. Tong, M. H. Wang, Z. F. Zheng and Y. Qin, *Angew. Chem., Int. Ed.*, 2017, **56**, 816–820.

16 J. N. Hui, G. Zhang, C. S. Ni and J. T. S. Irvine, *Chem. Commun.*, 2017, **53**, 10038–10041.

17 A. T. Garcia-Esparza, T. Shinagawa, S. Ould-Chikh, M. Qureshi, X. Y. Peng, N. N. Wei, D. H. Anjum, A. Clo, T. C. Weng, D. Nordlund, D. Sokaras, J. Kubota, K. Domen and K. Takanabe, *Angew. Chem., Int. Ed.*, 2017, **56**, 5780–5784.

18 Q. Wang, T. Hisatomi, Q. X. Jia, H. Tokudome, M. Zhong, C. Z. Wang, Z. H. Pan, T. Takata, M. Nakabayashi, N. Shibata, Y. B. Li, I. D. Sharp, A. Kudo, T. Yamada and K. Domen, *Nat. Mater.*, 2016, **15**, 611–615.

19 Y. J. Luo, X. M. Liu, X. H. Tang, Y. Luo, Q. Y. Zeng, X. L. Deng, S. L. Ding and Y. Q. Sun, *J. Mater. Chem. A*, 2014, **2**, 14927–14939.

20 M. Xiao, B. Luo, M. Q. Lyu, S. C. Wang and L. Z. Wang, *Adv. Energy Mater.*, 2018, **8**, 1701605.

21 Y. L. Wang, T. Nie, Y. H. Li, X. L. Wang, L. R. Zheng, A. P. Chen, X. Q. Gong and H. G. Yang, *Angew. Chem., Int. Ed.*, 2017, **56**, 7430–7434.

22 Y. G. Chen, S. Zhao, X. Wang, Q. Peng, R. Lin, Y. Wang, R. A. Shen, X. Cao, L. B. Zhang, G. Zhou, J. Li, A. D. Xia and Y. D. Li, *J. Am. Chem. Soc.*, 2016, **138**, 4286–4289.

23 Q. C. Yuan, D. Liu, N. Zhang, W. Ye, H. X. Ju, L. Shi, R. Long, J. F. Zhu and Y. J. Xiong, *Angew. Chem., Int. Ed.*, 2017, **56**, 4206–4210.

24 L. Shang, B. Tong, H. J. Yu, G. I. N. Waterhouse, C. Zhou, Y. F. Zhao, M. Tahir, L. Z. Wu, C. H. Tung and T. R. Zhang, *Adv. Energy Mater.*, 2016, **6**, 1501241.

25 G. X. Zhao, Y. B. Sun, W. Zhou, X. K. Wang, K. Chang, G. G. Liu, H. M. Liu, T. Kako and J. H. Ye, *Adv. Mater.*, 2017, **29**, 1703258.

26 K. F. Wu, Z. Y. Chen, H. J. Lv, H. M. Zhu, C. L. Hill and T. Q. Lian, *J. Am. Chem. Soc.*, 2014, **136**, 7708–7716.

27 Y. Y. Zhang, L. L. Han, C. H. Wang, W. H. Wang, T. Ling, J. Yang, C. K. Dong, F. Lin and X. W. Du, *ACS Catal.*, 2017, **7**, 1470–1477.

28 H. M. Huang, B. Y. Dai, W. Wang, C. H. Lu, J. H. Kou, Y. R. Ni, L. Z. Wang and Z. Z. Xu, *Nano Lett.*, 2017, **17**, 3803–3808.

29 X. J. Sun, D. D. Yang, H. Dong, X. B. Meng, J. L. Sheng, X. Zhang, J. Z. Wei and F. M. Zhang, *Sustainable Energy Fuels*, 2018, **2**, 1356–1361.

30 J. S. Zhang, M. W. Zhang, R. Q. Sun and X. C. Wang, *Angew. Chem., Int. Ed.*, 2012, **51**, 10145–10149.

31 X. C. Wang, X. F. Chen, A. Thomas, X. Z. Fu and M. Antonietti, *Adv. Mater.*, 2009, **21**, 1609–1612.

32 G. G. Zhang, M. W. Zhang, X. X. Ye, X. Q. Qiu, S. Lin and X. C. Wang, *Adv. Mater.*, 2014, **26**, 805–809.

33 Y. J. Cui, Z. X. Ding, X. Z. Fu and X. C. Wang, *Angew. Chem., Int. Ed.*, 2012, **51**, 11814–11818.

34 J. S. Zhang, G. G. Zhang, X. F. Chen, S. Lin, L. Mäglmann, G. Dolega, G. Lipner, M. Antonietti, S. Blechert and X. C. Wang, *Angew. Chem., Int. Ed.*, 2012, **51**, 3183–3187.

35 Z. Z. Lin and X. C. Wang, *Angew. Chem., Int. Ed.*, 2013, **52**, 1735–1738.

36 J. H. Sun, J. S. Zhang, M. W. Zhang, M. Antonietti, X. Z. Fu and X. C. Wang, *Nat. Commun.*, 2012, **3**, 1139.

37 Y. Zhou, Z. P. Tian, Z. Y. Zhao, Q. Liu, J. H. Kou, X. Y. Chen, J. Gao, S. C. Yan and Z. G. Zou, *ACS Appl. Mater. Interfaces*, 2011, **3**, 3594–3601.

38 X. X. Guo, X. Y. Qin, Z. J. Xue, C. B. Zhang, X. H. Sun, J. B. Hou and T. Wang, *RSC Adv.*, 2016, **6**, 48537–48542.

39 Y. X. Pan, Y. You, S. Xin, Y. T. Li, G. T. Fu, Z. M. Cui, Y. L. Men, F. F. Cao, S. H. Yu and J. B. Goodenough, *J. Am. Chem. Soc.*, 2017, **139**, 4123–4129.

40 L. Liang, F. C. Lei, S. Gao, Y. F. Sun, X. C. Jiao, J. Wu, S. Qamar and Y. Xie, *Angew. Chem., Int. Ed.*, 2015, **54**, 13971–13974.

41 A. Etogo, R. Liu, J. B. Ren, L. W. Qi, C. C. Zheng, J. Q. Ning, Y. J. Zhong and Y. Hu, *J. Mater. Chem. A*, 2016, **4**, 13242–13250.

42 R. Jinnouchi, A. V. Akimov, S. Shirai, R. Asahi and O. V. Prezhdo, *J. Phys. Chem. C*, 2015, **119**, 26925–26936.

43 P. Kwolek, K. Pilarczyk, T. Tokarski, K. Lewandowska and K. Szacilowski, *Nanoscale*, 2014, **6**, 2244–2254.

44 W. J. Fang, Z. Jiang, L. Yu, H. Liu, W. F. Shangguan, C. Terashima and A. Fujishima, *J. Catal.*, 2017, **352**, 155–159.

45 H. Li, F. Qin, Z. P. Yang, X. M. Cui, J. F. Wang and L. Z. Zhang, *J. Am. Chem. Soc.*, 2017, **139**, 3513–3521.

46 D. W. Li, S. X. Ouyang, H. Xu, D. Lu, M. Zhao, X. L. Zhang and J. H. Ye, *Chem. Commun.*, 2016, **52**, 5989–5992.

47 D. D. Zheng, X. N. Cao and X. C. Wang, *Angew. Chem., Int. Ed.*, 2016, **55**, 11512–11516.

48 Y. F. Xu, M. Z. Yang, B. X. Chen, X. D. Wang, H. Y. Chen, D. B. Kuang and C. Y. Su, *J. Am. Chem. Soc.*, 2017, **139**, 5660–5663.

49 J. X. Low, J. G. Yu, M. Jaroniec, S. Wageh and A. A. Al-Ghamdi, *Adv. Mater.*, 2017, **29**, 1601694.

50 P. J. Yang, J. H. Zhao, J. Wang, B. Y. Cao, L. Li and Z. P. Zhu, *J. Mater. Chem. A*, 2015, **3**, 8256–8259.

51 J. H. Hu, L. J. Wang, P. Zhang, C. H. Liang and G. S. Shao, *J. Power Sources*, 2016, **328**, 28–36.

52 L. J. Zhang, S. Li, B. K. Liu, D. J. Wang and T. F. Xie, *ACS Catal.*, 2014, **4**, 3724–3729.

53 X. Yang, J. Xu, T. L. Wong, Q. D. Yang and C. S. Lee, *Phys. Chem. Chem. Phys.*, 2013, **15**, 12688–12693.

54 W. Y. Wang, X. W. Wang, C. X. Zhou, B. Du, J. X. Cai, G. Feng and R. B. Zhang, *J. Phys. Chem. C*, 2017, **121**, 19104–19111.

55 S. B. Wang, B. Y. Guan and X. W. Lou, *J. Am. Chem. Soc.*, 2018, **140**, 5037–5040.

56 M. Lan, R. M. Guo, Y. B. Dou, J. Zhou, A. W. Zhou and J. R. Li, *Nano Energy*, 2017, **33**, 238–246.

57 K. E. deKrafft, C. Wang and W. B. Lin, *Adv. Mater.*, 2012, **24**, 2014–2018.

58 H. R. Chen, K. Shen, J. Y. Chen, X. D. Chen and Y. W. Li, *J. Mater. Chem. A*, 2017, **5**, 9937–9945.

59 Z. F. Huang, J. J. Song, K. Li, M. Tahir, Y. T. Wang, L. Pan, L. Wang, X. W. Zhang and J. J. Zou, *J. Am. Chem. Soc.*, 2016, **138**, 1359–1365.

60 J. M. Chen, J. Y. Chen and Y. W. Li, *J. Mater. Chem. A*, 2017, **5**, 24116–24125.

61 S. B. Wang, B. Y. Guan, Y. Lu and X. W. Lou, *J. Am. Chem. Soc.*, 2017, **139**, 17305–17308.

62 X. X. Zhao, J. R. Feng, J. Liu, W. Shi, G. M. Yang, G. C. Wang and P. Cheng, *Angew. Chem., Int. Ed.*, 2018, **57**, 9790–9794.

63 L. He, T. E. Wood, B. Wu, Y. C. Dong, L. B. Hoch, L. M. Reyes, D. Wang, C. Kübel, C. X. Qian, J. Jia, K. Liao, P. G. O'Brien, A. Sandhel, J. Y. Y. Loh, P. Szymanski, N. P. Kherani, T. C. Sum, C. A. Mims and G. A. Ozin, *ACS Nano*, 2016, **10**, 5578–5586.

64 B. L. Tao, Y. Zhang, D. Z. Han, Y. P. Li and Z. F. Yan, *J. Mater. Chem. A*, 2014, **2**, 5455–5461.

65 H. Huang, Z. K. Yue, G. Li, X. M. Wang, J. Huang, Y. K. Du and P. Yang, *J. Mater. Chem. A*, 2014, **2**, 20118–20125.

66 D. F. Wang, Z. G. Zou and J. H. Ye, *Chem. Mater.*, 2005, **17**, 3255–3261.

67 A. Nashim, S. Martha and K. M. Parida, *ChemCatChem*, 2013, **5**, 2352–2359.

68 S. J. Hu, B. Chi, J. Pu and L. Jian, *J. Mater. Chem. A*, 2014, **2**, 19260–19267.

69 S. Martha, K. H. Reddy and K. M. Parida, *J. Mater. Chem. A*, 2014, **2**, 3621–3631.

70 S. W. Cao, X. F. Liu, Y. P. Yuan, Z. Y. Zhang, Y. S. Liao, J. Fang, S. C. J. Loo, T. C. Sum and C. Xue, *Appl. Catal., B*, 2014, **147**, 940–946.

71 D. D. Ma, J. W. Shi, Y. J. Zou, Z. Y. Fan, J. W. Shi, L. H. Cheng, D. K. Sun, Z. Y. Wang and C. M. Niu, *Nanoscale*, 2018, **10**, 7860–7870.

72 B. C. Qiu, Q. H. Zhu, M. M. Du, L. G. Fan, M. Y. Xing and J. L. Zhang, *Angew. Chem., Int. Ed.*, 2017, **56**, 2684–2688.

73 J. R. Ran, X. L. Wang, B. C. Zhu and S. Z. Qiao, *Chem. Commun.*, 2017, **53**, 9882–9885.

74 K. Li, M. Han, R. Chen, S. L. Li, S. L. Xie, C. Y. Mao, X. H. Bu, X. L. Cao, L. Z. Dong, P. Y. Feng and Y. Q. Lan, *Adv. Mater.*, 2016, **28**, 8906–8911.

75 Z. Fang, J. J. Zhou, Y. M. Sun, J. H. Hu, L. Liang, R. Xu and H. W. Duan, *Nanoscale*, 2017, **9**, 17794–17801.

76 J. Feng, C. H. An, L. X. Dai, J. X. Liu, G. J. Wei, S. Bai, J. Zhang and Y. J. Xiong, *Chem. Eng. J.*, 2016, **283**, 351–357.

77 Q. M. Sun, N. Wang, J. H. Yu and J. C. Yu, *Adv. Mater.*, 2018, 1804368.

78 T. Simon, N. Bouchonville, M. J. Berr, A. Vaneski, A. Adrovic, D. Volbers, R. Wyrwich, M. Döblinger, A. S. Susha, A. L. Rogach, F. Jäckel, J. K. Stolarczyk and J. Feldmann, *Nat. Mater.*, 2014, **13**, 1013–1018.

79 L. Yuan, C. Han, M. Q. Yang and Y. J. Xu, *Int. Rev. Phys. Chem.*, 2016, **35**, 1–36.

80 Z. R. Tang, B. Han, C. Han and Y. J. Xu, *J. Mater. Chem. A*, 2017, **5**, 2387–2410.

81 M. Q. Yang, C. Han and Y. J. Xu, *J. Phys. Chem. C*, 2015, **119**, 27234–27246.

82 B. Han, S. Q. Liu, N. Zhang, Y. J. Xu and Z. R. Tang, *Appl. Catal., B*, 2017, **202**, 298–304.

83 Y. X. Pan, H. Q. Zhuang, J. D. Hong, Z. Fang, H. Liu, B. Liu, Y. Z. Huang and R. Xu, *ChemSusChem*, 2014, **7**, 2537–2544.

84 W. Cho, H. J. Lee and M. Oh, *J. Am. Chem. Soc.*, 2008, **130**, 16943–16946.

85 M. Y. Ye, Z. H. Zhao, Z. F. Hu, L. Q. Liu, H. M. Ji, Z. R. Shen and T. Y. Ma, *Angew. Chem., Int. Ed.*, 2017, **56**, 8407–8411.

86 X. F. Li, C. Q. Wang, B. Li, Y. Shao and D. Z. Li, *Phys. Chem. Chem. Phys.*, 2016, **18**, 27848–27857.

87 Y. T. Xiao, G. H. Tian, Y. J. Chen, X. Zhang, H. Y. Fu and H. G. Fu, *Appl. Catal., B*, 2018, **225**, 477–486.

88 F. Zhang, X. Y. Li, Q. D. Zhao and D. K. Zhang, *ACS Sustainable Chem. Eng.*, 2016, **4**, 4554–4562.

89 V. Amoli, M. G. Sibi, B. Banerjee, M. Anand, A. Maurya, S. A. Farooqui, A. Bhaumik and A. K. Sinha, *ACS Appl. Mater. Interfaces*, 2015, **7**, 810–822.

90 Z. Wang, W. W. Wu, Q. Xu, G. D. Li, S. H. Liu, X. F. Jia, Y. Qin and Z. L. Wang, *Nano Energy*, 2017, **38**, 518–525.

91 J. Z. Chen, X. J. Wu, L. S. Yin, B. Li, X. Hong, Z. X. Fan, B. Chen, C. Xue and H. Zhang, *Angew. Chem., Int. Ed.*, 2015, **54**, 1210–1214.

92 S. Y. Bao, Q. F. Wu, S. Z. Chang, B. Z. Tian and J. L. Zhang, *Catal. Sci. Technol.*, 2017, **7**, 124–132.

93 Z. W. Tong, D. Yang, Z. Li, Y. H. Nan, F. Ding, Y. C. Shen and Z. Y. Jiang, *ACS Nano*, 2017, **11**, 1103–1112.

94 S. N. Talapaneni, G. P. Mane, D. H. Park, K. S. Lakhi, K. Ramadass, S. Joseph, W. M. Skinner, U. Ravon, K. Al-Bahily and A. Vinu, *J. Mater. Chem. A*, 2017, **5**, 18183–18192.

95 N. Srinivasan, E. Sakai and M. Miyauchi, *ACS Catal.*, 2016, **6**, 2197–2200.