



Cite this: *Chem. Soc. Rev.*, 2016,  
45, 5408

## Metal–organic frameworks as competitive materials for non-linear optics

L. R. Mingabudinova,<sup>a</sup> V. V. Vinogradov,<sup>a</sup> V. A. Milichko,<sup>a</sup> E. Hey-Hawkins<sup>\*b</sup> and A. V. Vinogradov<sup>\*a</sup>

The last five years have witnessed a huge breakthrough in the creation and the study of the properties of a new class of compounds – metamaterials. The next stage of this technological revolution will be the development of active, controllable, and non-linear metamaterials, surpassing natural media as platforms for optical data processing and quantum information applications. However, scientists are constantly faced with the need to find new methods that can ensure the formation of quantum and non-linear metamaterials with higher resolution. One such method of producing metamaterials in the future, which will provide scalability and availability, is chemical synthesis. Meanwhile, the chemical synthesis of organized 3D structures with a period of a few nanometers and a size of up to a few millimeters is not an easy task and is yet to be resolved. The most promising avenue seems to be the use of highly porous structures based on metal–organic frameworks that have demonstrated their unique properties in the field of non-linear optics (NLO) over the past three years. Thus, the aim of this review is to examine current progress and the possibilities of using metal–organic frameworks in the field of non-linear optics as chemically obtained metamaterials of the future. The review begins by presenting the theoretical principles of physical phenomena represented by mathematical descriptions for clarity. Major attention is paid to the second harmonic generation (SHG) effect. In this section we compare inorganic single crystals, which are most commonly used to study the effect in question, to organic materials, which also possess the required properties. Based on these data, we present a rationale for the possibility of studying the non-linear optical properties of metal–organic structures as well as describing the use of synthetic approaches and the difficulties associated with them. The second part of the review explicitly acquaints the reader with a new class of materials which successfully combines the positive properties of organic and inorganic materials. Using recently synthesized metal–organic frameworks and coordination polymers in the field of non-linear optics as an example, we consider synthetic approaches used for obtaining materials with desired properties and the factors to be considered in this case. Finally, probable trends towards improving the quality of the synthesized materials with regards to their further use in the field of non-linear optical effects are described.

Received 21st May 2016

DOI: 10.1039/c6cs00395h

[www.rsc.org/chemscrev](http://www.rsc.org/chemscrev)

## 1. Introduction

### 1.1 A brief introduction to non-linear optics

Before proceeding to the consideration of current trends in the field of materials used for non-linear optics, one should be familiar with the existing theoretical propositions in order to have an understanding of the processes observed in the studies of the corresponding effects.

When considering the interaction of a light wave with matter, we introduce the polarization vector  $P$ , which is a function of the strength of the electric field of the light wave:  $P = f(E)$ . At a time

when studying the interaction of radiation with matter involved light sources with a small capacity and providing an intensity of the order of  $1\text{--}10 \text{ W cm}^{-2}$  and achievable light fields of  $0.1\text{--}10 \text{ V cm}^{-1}$  as a result, it was believed that the investigated experimental characteristics did not depend on the radiation intensity. The proposed assertion can be attributed to the field of linear optics, where the dielectric constant  $\epsilon$  and the magnetic susceptibility  $\chi$  are treated as independent of the intensity of electromagnetic waves propagating in the medium, which corresponds to a linear relationship between the polarization vector  $P$  and the electric field strength  $E$  (Fig. 1(a)).<sup>1</sup> However, with the advent of laser sources with an intensity of  $10^{10} \text{ W cm}^{-2}$  and the resulting field of about  $10^7 \text{ V cm}^{-1}$ , one cannot neglect an intra-atomic field which is of the order of  $e/r_o^2 \sim 10^9 \text{ V cm}^{-1}$  ( $r_o$  is the characteristic radius of an electronic orbital, which is equal to a value of about  $10^{-8} \text{ cm}$ ).

<sup>a</sup> ITMO University, St. Petersburg, 197101, Russian Federation.

E-mail: [avv@scam.ru](mailto:avv@scam.ru)

<sup>b</sup> Leipzig University, Faculty of Chemistry and Mineralogy, Institute of Inorganic Chemistry, D-04103 Leipzig, Germany. E-mail: [hey@uni-leipzig.de](mailto:hey@uni-leipzig.de)



In such an analysis, an equation relating the electric field strength and polarization can be generally represented by expanding as a series:

$$P = \kappa E + \chi EE + \vartheta EEE + \dots,$$

where the first term represents the linear component, and the other terms belong to the non-linear polarization components. Such a presentation of the expansion equation is rather conditional, since the values of  $\kappa$ ,  $\chi$ ,  $\vartheta$  are tensors of the 2nd, 3rd and higher ranks.

Changing the degree of intensity and exposure time also affects the change of the light wave field over time  $E(t) = E_0 \cos(\omega t)$ . The second harmonic originates from the quadratic term  $\chi EE$  of



**L. R. Mingabudinova**

*Dr. Evamarie Hey-Hawkins, Leipzig, Germany. Her current research interests include the synthesis of new MOFs and their biological applicability and non-linear optical effects.*

*Leila Mingabudinova received her BSc degree from Saint Petersburg State University, Saint Petersburg, Russia, in 2015. She has prepared her Master's thesis at the International Laboratory "Solution Chemistry of Advanced Materials and Technologies", led by Alexandr Vinogradov, ITMO University, Saint Petersburg, Russia, and partially at Leipzig University, Faculty of Chemistry and Mineralogy, Institute of Inorganic Chemistry, led by Prof.*

the polarization  $P$  expansion in powers of the electric field of the light wave. The material polarization component, which occurs in the second term of the equation, can be expressed as follows:

$$P_{(2)} = 1/2\epsilon_0(\chi(2)E_0^2) + (\chi^2E_0^2 \cos(2\omega t))$$

It can be seen that polarization has a single permanent component and another component at a frequency of  $2\omega$ . The value of  $\chi$  is the quadratic non-linear polarizability of a substance. The prerequisite for the second harmonic generation (SHG, doubling of the incident light frequency) is an inequality of  $\chi$  to zero. These conditions are achieved in anisotropic substances that do not have a center of symmetry. In the case of a medium being isotropic or having a center of symmetry, a change in the



**V. V. Vinogradov**

*University of Jerusalem. In 2014, he joined ITMO University, where he started as a Head of the Laboratory of Solution Chemistry of Advanced Materials & Technologies.*

*Vladimir Vinogradov was born in Ivanovo, Russia, in 1985. He received a MS Degree in materials science from the Ivanovo State University of Chemistry and Technology in 2007. He was awarded a PhD in inorganic chemistry from the Institute of Solution Chemistry of RAS (with Professor Alexander V. Agafonov) in 2010. Thereafter, he did postdoctoral training in materials science in Prof. David Avnir's research group at the Hebrew*



**V. A. Milichko**

*Valentin A. Milichko is an Assistant Professor at ITMO University, Department of Nanophotonics and Metamaterials. He received his PhD in laser physics from the Institute of Automation and Control Processes, Russian Academy of Sciences, in 2014 and was a postdoctoral member of ITMO University during 2014–2016. His research focuses on the non-linear optics of metal-organic solids and biomolecules.*



**E. Hey-Hawkins**

*Evamarie Hey-Hawkins received her doctoral degree (1983) at the University of Marburg, Germany. After stays at the University of Sussex, UK, the University of Western Australia and the ANU, Canberra, Australia, she returned to Germany and completed her habilitation (Marburg, 1988). From 1988 to 1990 she was a Research Associate at the Max Planck Institute for Solid State Research, Stuttgart, Germany, then a Heisenberg fellow at Technical University Karlsruhe, Germany (1990–1993). Since 1993, she has a Chair in Inorganic Chemistry at Universität Leipzig, Germany. Since October 2013, she has been the Chair of the COST Action "Smart Inorganic Polymers" (CM1302, SIPS, [www.sips-cost.org](http://www.sips-cost.org)). Her scientific interests are in the field of organophosphorus chemistry, biologically active boron compounds and transition metal complexes for applications in catalysis and materials science.*



direction of the applied electric field  $E$  would lead to a change in the sign of the polarization  $P$ , which implies that the components of the equation containing even powers in the expansion should not contribute to the total value, *i.e.*, should be equal to zero. If the frequencies are doubled, a medium should have no absorption for all interacting waves. Consequently, the second harmonic generation is determined by the symmetry of the crystal structure; therefore, compounds without a center of inversion are suitable for studying this effect.

## 1.2 Phase synchronism conditions

Interaction between light beams ( $\omega_1$  and  $\omega_2$ ) inside matter can result in the generation of beams with a sum (or difference) frequency ( $\omega_3$ ). The most effective case when the resulting frequency  $\omega_3$ , attributable to the given element of volume from the preceding elements, corresponds to the desired phase of the radiation at the same frequency generated in this volume element. In this case, the intensity of generation will increase by several orders of magnitude, since the process of accumulation takes place throughout the non-linearity of the medium, which is fulfilled under the conditions of phase (wave and spatial) synchronism between beams' wave vectors:

$$k_1 + k_2 = k_3,$$

where  $k$  is a wave vector.

Second harmonic generation is a special case of the total frequency generation  $\omega_1 + \omega_2 = \omega_3$ , in which waves of the same frequency enter a medium, and the total radiation is  $\omega + \omega = 2\omega$ . The conditions of phase synchronism for the case when the wave vectors are collinear can be written in terms of wave numbers as follows:

$$K_\omega + k_\omega = k_{2\omega}$$

when the wave propagates in a medium with a refractive index  $n$ , then  $k = 2\pi n/\lambda$ .

Taking into account the phase synchronism, we obtain  $n_\omega + n_\omega = 2n_{2\omega}$ . Thus, it is necessary to keep the condition  $n_\omega = n_{2\omega}$  in a non-linear medium.



A. V. Vinogradov

Alexandr V. Vinogradov obtained his PhD degree from the Ivanovo State University of Chemistry and Technology (ISUCT) in 2011. He spent some time as a postdoc at Leipzig University, Germany, in the group of Prof. Evamarie Hey-Hawkins. In 2014, he joined ITMO University, where he is currently associate professor. His specific interests are in the field of bioapplication and optical properties of MOFs, photoactive thin films and inkjet printing of optical nanostructures.

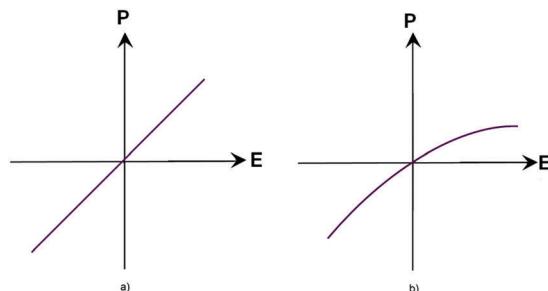


Fig. 1 Relationship between the  $E$  and  $P$  vectors in a linear medium (a) and in a non-linear medium (b).

Typically, in media with normal dispersion, such a condition is not fulfilled, since the function  $n(\omega)$  increases with increasing frequency. There are several ways to overcome it. Among them, the use of crystals with birefringence has the greatest practical importance. In such a crystal, one chooses a propagation direction for which the refractive index of an ordinary ray with the main frequency is equal to the refractive index of an extraordinary ray of the second harmonic. Fig. 2 schematically shows a cross-section of refractive index surfaces (refractive index indicatrix) for a negative uniaxial crystal, *i.e.* that for which at a given fixed frequency  $n_e < n_o$  (Fig. 2). The current values of extraordinary refractive indices (at an arbitrary angle  $\Theta$  between the wave vector and the optical axis of the crystal) are indicated by the superscript "e":  $n^e$ .

There are several classes of crystals lacking a center of symmetry for which  $n_{1o}(\omega) = n_2^e(\theta)|_{2\omega}$  in the case of propagation of waves with frequencies  $\omega$  and  $2\omega$  at the angle  $q_c$  with respect to the optical axis of the crystal. The angle between the optical axis of a non-linear crystal and the wave vectors of the interacting waves is called the phase (wave and spatial) synchronism angle.

As an example of practical application, Fig. 3 demonstrates several configurations for second harmonic generation in bulk materials and in waveguides, in which infrared light is converted into visible light and visible light into ultraviolet radiation.

According to Neumann's principle, the symmetry of a physical property must coincide with the crystal symmetry, so centrosymmetric crystals cannot possess the properties described by polar tensors of odd ranks. Thus, centrosymmetric crystals cannot have pyro- and piezoelectric, linear electrooptical properties, optical activity, *etc.*

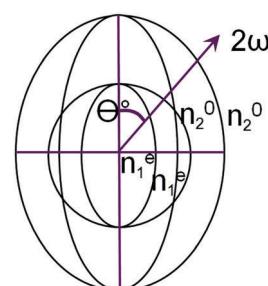


Fig. 2 The conditions of synchronism in a uniaxial crystal.



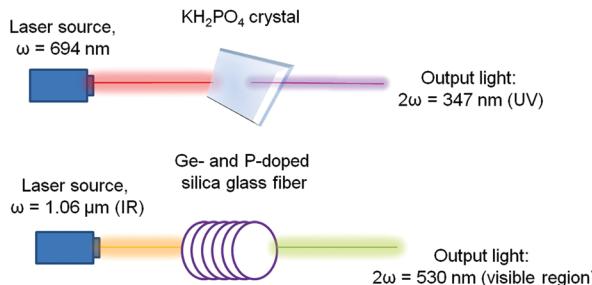


Fig. 3 Demonstration of the second harmonic in the visible region using different types of inorganic crystals. Adopted from ref. 1 (Copyright © 2000–2016 by John Wiley & Sons, Inc.).

The above fact is used to quickly examine the presence of second harmonic generation upon irradiation with a laser beam. Hence, the existence of the piezoelectric effect and second harmonic generation allows for an unequivocal conclusion that a substance is non-centrosymmetric; if the effect is expected but not observed, then the effect is apparently just too weak.

Observation of electrooptical and magnetooptical effects (dependence of the refractive index on static electric or magnetic fields) corresponds to the case of quadratic non-linearity if one of the fields is permanent. These effects are significant in an important class of photovoltaic and photorefractive effects. For example, in a number of semiconductors the influence of radiation results in a spatial separation of oppositely charged carriers (electrons and holes). This gives rise to an electrostatic field with a change in the permittivity tensor.

One of the major limitations in practical terms is that there are hardware limitations for studying non-linear optical properties of a material. Consequently, for many years research was mainly focused on the measurements of macro-objects as the transition to the micro level was limited by this problem. However, it is the field of micro-objects that will undergo further development of non-linear optics as successfully demonstrated in several studies using synthesized materials as an example. Switching to the said level is also interesting with regards to the fact that it is possible to make an assumption where structural units are largely responsible for the observed effect. In addition to this, it is worth noting that the material should also be resistant to laser radiation within a predetermined range to avoid thermal degradation. Thus, attention should turn to synthetic approaches, which facilitate obtaining the desired structure, and then to techniques for modifying the material to render it suitable for the process of investigation.

### 1.3 The appearance of NLO in different media

Since the attention of scientists has been focused on investigating various non-linear optical effects, entire classes of materials have been studied. However, it is worth noting that every material has its drawbacks. On the way to obtaining a material devoid of these deficiencies, the attention was shifted to organic materials. However, they also have failed to fully satisfy technical needs due to their high instability. Thus, the present time is characterized by a growing interest in hybrid materials,

which successfully combine the properties of both organic and inorganic components.

**1.3.1 Inorganic materials.** The non-centrosymmetric structure of inorganic crystals allows them to be used in investigations in the field of non-linear optics. The species particularly suitable for measuring characteristics are potassium dihydrogen phosphate, barium  $\beta$ -borate, and lithium niobate. For these crystals, values of  $\chi$  from 1 to  $100 \times 10^{-9}$  ESU were obtained due to the high electronic polarizability associated with the band structure of the crystal and its asymmetry.

Investigations on non-linear optical properties of the low-temperature modification  $\beta$ -Ba<sub>2</sub>B<sub>4</sub>O<sub>7</sub> (BBO) of barium borate started in the 1980s. The lack of a center of symmetry in this crystal provides a number of interesting properties, such as second harmonic generation with high efficiency (approximately 6 times greater than that of potassium dihydrogen phosphate), a high laser damage threshold, and large birefringence.<sup>2,3</sup> Single crystals of BBO found applications in optoelectronics for conversion of the laser light frequency according to the mechanism of generation of second and higher harmonics. Electronic delocalization in complex boron–oxygen anions leads to the presence of non-linear optical properties in their single crystals.<sup>4–7</sup> The high polarizability of borate crystals caused by the peculiarities of their electronic structure is comparable to the polarizability of organic aromatic compounds.<sup>8,9</sup>

The nature of the chemical bonds in the system (Fig. 4) is shown for a better understanding of non-linear optical properties in further studies of single crystals using metal–organic frameworks as an example. The  $sp^2$  hybridized boron in cyclic anions (Fig. 4a and b) interacts with an oxygen atom, providing a vacant orbital for the filled orbitals of the oxygen atoms, resulting in the formation of a  $\pi$ -bond within the ring. The interaction between the barium cations and borate anions is electrostatic; there is an intermolecular interaction<sup>10</sup> between electrically neutral layers.

The mechanical properties and resistance to laser radiation present in inorganic crystals are typical of these types of single crystals as well. Single crystals of borates are usually grown using the Czochralski method<sup>10</sup> or by direct recrystallization of the cooled melt. However, a tendency towards vitrification significantly complicates the process of growth. Moreover, low yields

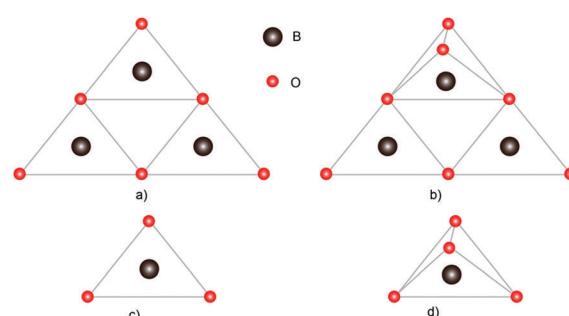


Fig. 4 Structural units that make up borate crystals: (a)  $B_3O_6$ , (b)  $B_3O_7$ , (c)  $BO_3$ , and (d)  $BO_4$ . Adapted from ref. 10 (with permission from Copyright © Royal Society of Chemistry 2016).



of suitable materials, undesirable variations in crystal properties, and energy consumption of production processes, ultimately lead to high costs of the finished optical elements.

Another inorganic crystalline compound used in non-linear optics is potassium dihydrogen phosphate,  $\text{KH}_2\text{PO}_4$  (KDP). This is a single-axis, negative piezoelectric crystal with the symmetry class  $\bar{4}2m$ . Its advantages include high radiation resistance (it is not destroyed when exposed to extremely high intensities of light radiation), relatively low level of absorption in the transparent region, and weak temperature dependence  $n(T)$  of the refractive index. However, there is a serious drawback in its practical application: the hygroscopy of the crystals, which requires certain measures when working with this material. Among the second class of inorganic crystals one can distinguish lithium niobate,  $\text{LiNbO}_3$ , another uniaxial negative piezoelectric crystal with the symmetry class  $3m$  and a transparency range of 0.4–5  $\mu\text{m}$ . The refractive indices of the crystal greatly depend on the temperature  $n(T)$  of the sample.

This crystal is grown by pulling from the melt. It is resistant to moisture but its linear dimensions with the necessary optical quality are substantially smaller than those of KDP. The radiation resistance is not inferior to the latter.

The quality factor for lithium niobate is higher than that of KDP, but a fairly high temperature dependence of the refractive index  $n(T)$  and a slightly higher level of absorption in the transparent region lead to a breach of the phase synchronism conditions. This is due to the rapid heating of the crystal under the action of radiation with extremely high intensity, such as radiation using fusion devices. However, due to this temperature dependence, lithium niobate allows for convenient 90° tangent SHG synchronism for radiation with a wavelength of about 1  $\mu\text{m}$  upon thermostating it at a temperature of 70–90 °C.

The optical quality of the crystals during growth may be impaired as a result of morphological instability, impurity contents, inclusions in the structure (due to the peculiarities of synthesis), viscosity, and tendency to vitrification for some systems.<sup>10</sup>

At the same time, the processes of growing such crystals are usually difficult and extremely energy intensive. The synthesis is carried out under high vacuum or in an inert atmosphere at a pressure of up to 1700 atm, and also requires a temperature ranging from 600 to 900 °C. In addition, it may take 10 days to 8 weeks to obtain a crystal of the size 20 mm × 20 mm × 60 mm. Moreover, of utmost importance is the fact that these methods do not allow the creation of a new design of structures, thereby limiting their potential use. Additional constraints are imposed due to the low stability in the presence of atmospheric moisture.

**1.3.2 Organic materials.** Organic materials possess high non-linear optical susceptibility due to increased polarizability of the  $\pi$ -electron cloud in conjugated systems. Upon applying small fields, the arising charge displacement leads to a non-linear response, which in turn leads to a modulation of the refractive index. In addition, polarization being purely electronic in nature has a response time of about one femtosecond or less.

In organic materials, each molecule should be considered as a separate dipole with non-linear polarization, and if the

distance between the molecules is much smaller than the wavelength of the incident light, the volumetric non-linear coefficient can be considered as the sum of dipoles (in non-coherent regime) with some correction ( $F$ ), taking into account the mutual orientation of the molecules.<sup>1</sup> Thus, the bulk values of non-linear coefficients  $\chi^{(2)}$  and  $\chi^{(3)}$  can be expressed in terms of corresponding molecular coefficients  $\beta$  and  $\gamma$ . In this case, we obtain expressions in the form:

$$\chi^{(2)} = N\beta F \quad \text{and} \quad \chi^{(3)} = N\gamma F.$$

How is it possible to increase the molecular non-linear susceptibility  $\beta$  and  $\gamma$  without specific physical treatment like coherent excitation? From the theory of a two-level model of a molecule (the ground state is  $g$ , excited is  $n$ ) it follows that the value of  $\beta$  depends on the oscillator strength ( $f$ ) for the  $g-n$  transition, frequency  $h_{\omega 0}$ , and a change in dipole moment in the transition from the ground state to the excited state,  $\Delta\mu$ . The donor-acceptor molecule of *p*-nitroaniline (*p*-na) can serve as an example of a molecule widely used in non-linear optics. The donor here is the  $\text{NH}_2$  group, the acceptor is the  $\text{NO}_2$  group, and both groups are linked with the conjugated carbon system (namely, the aromatic ring) (Fig. 5).

To produce molecules with large values of  $\beta$ , one should select the strongest donor ( $-\text{NH}_2$ ,  $-\text{NMe}_2$ ) and acceptor ( $-\text{CN}$ ,  $-\text{NO}_2$ ) groups and increase the conjugated bond length between molecules. The transition from such dipole molecules to the bulk material with high non-linear polarizability  $\chi^{(2)}$  is investigated while considering the structure of the material. The problem is that in most cases of crystallization the dipole molecules line up in such a way that their dipole moments are oppositely directed. In this case, the total optical polarizability is close to zero. However, there are several ways to solve this problem. One is to coat thin ordered molecular layers using the Langmuir-Blodgett method, whereby monomolecular layers are applied sequentially so that the dipole moments of the molecules are directed in the same direction.

**1.3.3 Organic chromophores.** One of the most successful approaches to overcome the existing shortcomings in inorganic materials is the use of organic chromophores as an alternative.

Scientists have achieved impressive results in this area. However, the creation of materials with directional optical properties remains an extremely difficult task to solve. Therefore, the majority of approaches to the preparation of materials based on chromophores rely purely on the chemical intuition of the researchers. In recent decades, heterocyclic fragments have

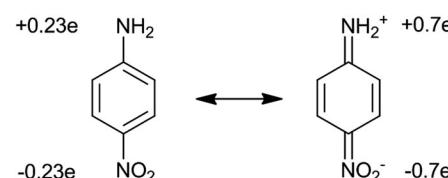


Fig. 5 Molecular structures for the ground and excited states and the degree of charge transfer from donor to acceptor in the ground and excited states.



been considered to be promising for chromophores, in particular as elements for the  $\pi$  conjugation. Moreover, theoretical calculations have already been performed for chromophores possessing one or more thiophene as well as triazole, pyrrole or furan rings.

To achieve high coefficients in non-linear optical properties, an optimal choice of the electron donor/acceptor in the system is necessary, which also poses a challenge for researchers at the moment. In recent years, major attention has been focused on electron bridges in the chromophore structure and the properties of the acceptor with no emphasis on the properties of the donor, although it also affects the final characteristics of the system. The class of alkyl and aryl amines, which have a lone pair of electrons on nitrogen suitable for interacting with the  $\pi^*$  orbital of the aromatic ring, turned out to be the most successful for applications. Perhaps this was the reason that prompted scientists to pay attention to the introduction of additional heteroatoms in the fragments of the chromophore donor, which sharply modulates conjugation of the  $\pi$ -electron system and changes the kinetics of charge transfer, affecting in turn the first-order hyperpolarizability of the chromophores. Moreover, the introduction of heteroatoms in the structure allows for further synthetic modification. However, it should be borne in mind that in this case the problem of obtaining a quality product as a result of a multi-stage chemical synthesis still persists.

Examples of such solutions could be provided by the creation of a series of chromophores<sup>11</sup> with one or two heteroatoms in the *ortho* position relative to the  $\pi$  bridge, where more efficiently conjugation of these heteroatoms with the  $\pi$  system may occur. For correct comparison of the properties of chromophores, 2-dicyanomethylene-3-cyano-4,5,5-dimethyl-2,5-dihydrofuran (tcf) can be used as an electron acceptor in each system.

Thus, due to the presence of the strong donor group (*E*)-2-(4-(2,4-bis(diethylamino)styryl)-3-cyano-5,5-dimethylfuran-2(5*H*)-ylidene)malononitrile, this chromophore type showed the greatest polarizability in the series compared to the rest. The chromophore showed a small dihedral angle 3.10°, whereas 3-(*E*)-2-(3-cyano-4-(2-ethoxy-4,6-di(pyrrolidin-1-yl)styryl)-5,5-dimethylfuran-2(5*H*)-ylidene)malononitrile, which showed the lowest hyperpolarizability, has a corresponding angle of 22°. Thus, a low value of  $\beta$  is observed due to a large torsion angle between the acceptor group and the benzene ring, which prevents effective  $\pi$  conjugation.

The obvious disadvantages of organic chromophores are multistage syntheses, the need for product purification after each stage to avoid isomerization of initial compounds in an undesired phase, very low yields of the isolated product, low solubility in common solvents used for the crystallization of the material, and the aggregation of some of the compounds in the solution, which again complicates the process of separation and purification of the final product. Ultimately, everything leads to a significant loss of material and an increase in its cost.<sup>11</sup>

For structures with large molecular hyperpolarizability, the system requires the presence of electrons with a highly efficient response to the electric field, and asymmetry at the molecular level. Increased polarizability is provided by  $\pi$ - $\pi$  interaction

between atoms in unsaturated fragments of a molecule. In this case, the electrons are delocalized and easily move under the influence of electric fields. Electron donor and acceptor groups located at opposite ends of a molecule are responsible for asymmetry (pull-push effect). These groups provide a permanent dipole moment, causing the induction of a total dipole moment, which ultimately leads to the asymmetry of the system. Increasing the strength of the donor/acceptor and the length of the conjugated chain between these groups leads to an increase in the hyperpolarizability, which is similar to what happens for the first-order dipole moment for two static charges.

For use in non-linear optics, the optically inactive polymer is doped with a chromophore employing mechanical methods (deposition). However, such a technique does not provide an ordered orientation of the chromophore; the result of which is a value of zero for  $\chi^2$ . To solve this problem, a tape with built-in chromophore molecules is heated above the temperature of vitrification (in most cases from 100 to 200 °C), thereby providing some degree of orientational mobility for the chromophore molecules. For the purpose of orientation of the molecules in the direction of the electric field, strong fields of the order of  $10^5$ – $10^6$  V m<sup>-1</sup> are used. The order in the system is then “frozen” by lowering the temperature. This approach yields a value of  $10^{-8}$  to  $10^{-7}$  ESU for  $\chi^2$ .<sup>12</sup> The main problem is instability over time due to randomization of the chromophore molecules. A sharp decay is possible at the beginning of the use of certain systems at room temperature even on the first day. However, this problem can be somewhat reduced by crosslinking polymers in which photoactive groups can form bonds upon ultraviolet irradiation thus preventing orientational mobility of the polymer. Crosslinked polymers in this case demonstrate a smaller  $\chi^2$  than non-crosslinked ones, but the decay rate is greatly reduced.<sup>13</sup> In addition, it is possible to increase the stability of chromophores through the use of polyimides in order to reduce orientational mobility of the polymer structure.<sup>14,15</sup> It is also possible to use polyamic acid which can be attached as a pendant group. When heated above 250 °C, the compound undergoes condensation with the formation of an imide ring, which limits the allowable motion of chromophores and increases the thermal stability of the material. Then, the process of sealing the film is carried out at a temperature of about 350 °C, which reduces the film thickness to 40%,<sup>14</sup> further limiting the mobility of the guest molecules.

It should be noted that despite the fact that thermal treatment increases the thermal stability of the resulting film, it is important to bear in mind that the decay temperature for the majority of non-linear chromophores is significantly lower than the temperature required for processing.

#### 1.4 Structurally organized materials

**1.4.1 Langmuir–Blodgett (LB) films.** In the section on organic materials with NLO properties, we have already revealed the prospects of using Langmuir–Blodgett films.

They consist of molecules, one end of which has a hydrophilic group and the other a hydrophobic group distributed on the surface, essentially forming a monolayer. This method



allows the thickness and structure of the final multilayer to be controlled by layer-by-layer deposition by dipping the substrate into a solution. For this purpose, the first stage involves the adsorption of molecules that are inactive from the perspective of non-linear optics, and the next stage the application of the chromophore molecules. This approach allowed obtaining non-linear susceptibility, which was approximately 20 times higher than that of barium  $\beta$ -borate.<sup>16</sup>

The main difficulty while applying films using the Langmuir–Blodgett method is in controlling the surface pressure in the adsorption process and in removing the molecules. Since it is pressure that is responsible for the dense packing of the molecules, its decrease may result in lowering of the orientation, whereas higher pressures may cause the molecules to fold in on each other, thereby destroying the layer. It is therefore necessary to maintain a constant pressure in the system.

At first glance, the technique is simple because the deposition process itself does not require a high temperature or an ultrahigh vacuum. But one should not forget that the very realization of the experiment requires a cleanroom. Other serious drawbacks of this advantageous method are the technical complexity of the procedure and greater sensitivity to environmental conditions. However, the most significant drawback is low mechanical and thermal stability, since the layers are connected to each other only by van der Waals forces. Additionally, upon increasing the heat over time, chromophore disorientation in the layers may occur.

Molecules with greater hyperpolarizability possess higher efficiency of charge transport in an optically excited state, resulting in a large difference between the ground and excited states of the dipole moments. These excitations involve  $\pi$ – $\pi^*$  transitions in one predominant direction.<sup>17</sup> It generally refers to the direction of the dipole moment of a molecule having a cylindrical symmetry. In LB films such a molecule is tilted at an angle  $\Phi$  with respect to the surface normal. It turns out that the molecules in the Langmuir–Blodgett films are highly oriented, causing a change in the local optical field. For example, it has been shown that interface nonlinearity is represented by a local (or electric dipole allowed) surface nonlinear susceptibility tensor.<sup>18,19</sup> This is due to the induced dipole–dipole interaction between the molecules.

Shen and coworkers<sup>20–22</sup> pioneered investigations of second harmonic generation in monolayers. The Heinz group<sup>23</sup> studied the resonant amplification of second harmonic generation from rhodamine dye molecules adsorbed on the substrate of molten silicon. By changing the polarization of the incident light and measuring the output polarization using the corresponding mathematical equation, they obtained a  $\Phi$  value of 30° and a hyperpolarization of  $30 \times 10^{-30}$  ESU.

Researchers from the Girling group<sup>24</sup> studied the processes of deposition of both monolayer and bilayer films of merocyanine dye for second harmonic generation. Upon exposing this dye to air, protonation occurred resulting in loss of non-linear optical efficiency. For this reason, experiments were performed in an ammonia atmosphere. The obtained values of  $\chi$  correspond to  $500 \times 10^{-30}$  ESU. Studies of non-linear optical multilayer

samples can also be observed for the Z-type films with, for example, polar films that are not found in nature, as the molecular dipoles prefer an antiparallel orientation. Another approach is to study Y-type films with effective molecules in terms of non-linear optics, which alternate either with non-active molecules or molecules whose hyperpolarizability has the opposite sign. In the latter case, despite the fact that the dipoles are arranged antiparallel, which is more advantageous in terms of energy, their hyperpolarizabilities cancel each other out.

Experiments with alternation of fatty acid molecules that are inactive in terms of optics proved to be ineffective in contrast to expectations. In this regard, Neal<sup>25</sup> proposed a different method of alternation of layers, where hemicyanine and aminonitrostilbene dyes were used, whose hyperpolarizabilities are opposite in sign. Thus, despite the fact that the deposition was performed by Y-type, the values of hyperpolarizabilities were added. It has been experimentally shown that an amplification of the second harmonic signal is observed as a result of the film formation. It has been determined that the  $\chi$  values for hemicyanine and aminonitrostilbene amount to  $300 \times 10^{-30}$  and  $55 \times 10^{-30}$  ESU, respectively. Evidence of interaction between the layers was provided by the obtained value of  $\chi$ ,  $850 \times 10^{-30}$  ESU for the bilayer, which was significantly larger than the values of the individual components. In continuation of a series of experiments, the second harmonic generation for films consisting of bilayers was investigated which led to interesting results. It turned out that the obtained dependence of intensity of the second harmonic generation on the number of layers was greater than quadratic, which may indicate a disorder of the multilayer.<sup>25</sup>

**1.4.2 Covalent self-assembly.** Systems of self-assembled monolayers, or so-called chemisorbed films, can cope with some drawbacks of the Langmuir–Blodgett technique. The method consists of immersing hydroxylated substrates in an amphiphilic liquid (solution or melt). Thus, covalent attachment to the surface occurs (Fig. 6). Then, new layers can be applied to the surface using chemical activation.

The main advantage of this technique is the ability to obtain the desired structural organization of the material. Such materials exhibit good chemical, thermal and electrical stability, and can even be additionally crosslinked to improve their properties for practical use. This technology is a convenient way of organizing non-centrosymmetry in the structure. Marks *et al.* implemented a strategy in which the  $-\text{SiCl}_3$  groups were attacked by small molecules and, in an  $\text{S}_{\text{N}}2$  reaction mechanism, interacted with molecules of an NLO-active dye.<sup>26</sup>

Self-assembled layers of [2-(*p*-chloromethylphenyl)ethynyl]-silane were reacted with [2-[4-[*N,N*-bis(3-hydroxypropyl)amino]phenyl]ethynyl]-4'-pyridine. Also, a system has been considered, produced by applying a thin layer of [[4-[*N,N*-bis(3-hydroxypropyl)amino]phenyl]azo]-4'-pyridin, by centrifugation on a substrate modified with benzyl chloride, and subsequent annealing at a temperature of 110 °C, which yielded a material with a high degree of organization (Fig. 7).

With this method it was possible to synthesize a three-layer arrangement in just one hour. The combined use of self-assembled



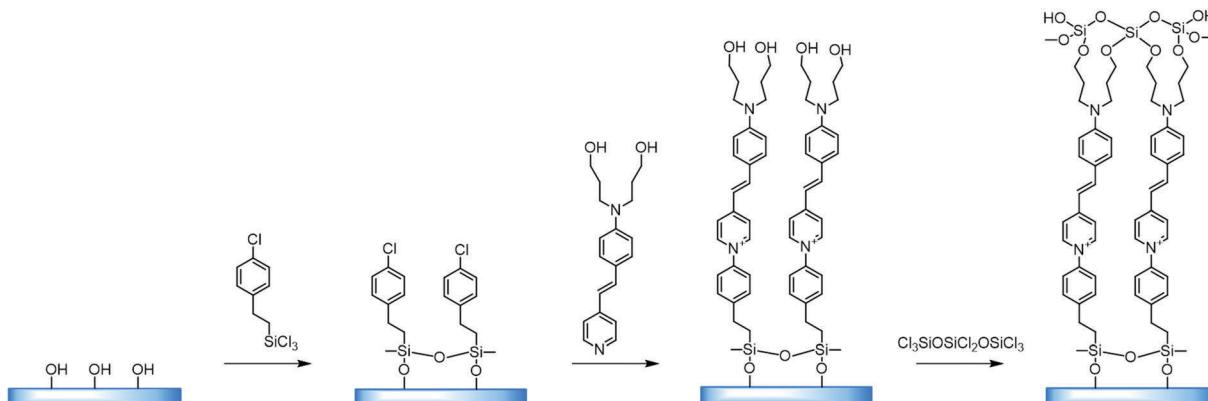


Fig. 6 An example of a covalent self-assembly process for (dialkylamino)stilbazole on the surface of benzyl chloride-modified silica. The third stage is the crosslinking between the hydroxyl groups and activation for applying the next layer. Adapted from ref. 29 (with permission from Copyright © 2016 American Chemical Society).

layers and deposition of thin films by means of centrifugation is a significant step on the way to application in electrooptical devices.

However, it is worth considering that the process may take time – from four hours to a few days.<sup>27</sup> Also, surface activation and adsorption processes require elevated temperatures that can cause disordering of the spatial orientation of the molecules.<sup>28</sup> Therefore, the maximum reactivity and organization of the surface layer are necessary factors.

Despite the technical difficulties of the process, some produced structures demonstrate a susceptibility which is more than 60 times higher than that observed for barium  $\beta$ -borate crystals.<sup>29</sup>

**1.4.3 Ionic self-assembled monolayer (ISAM) technique.** Compared to the method originally proposed in 1992 by Decher *et al.*,<sup>30</sup> this newly proposed method has a number of advantages. The method implies an emphasis on the forces of Coulomb interaction between oppositely charged polymers forming ultrathin layers. The process itself consists of alternating cycles of immersion of the substrate in a solution of a polyelectrolyte with an opposite charge. During immersion the ions bind to the surface. Some part of the bound groups causes recharge of the surface, resulting in limitation of the further attachment of ions. After removal, the substrate is washed with

an aqueous solution to remove the unbound polymer. The process is repeated until the desired film thickness is reached. It is not only possible to obtain a bilayer organization of the material, but also to alternate layers with different functional properties.<sup>31</sup>

## 1.5 Conclusions of the theoretical section

As indicated above, various materials used in the field of nonlinear optics have a number of disadvantages, which prompt the continuing development of new ways of obtaining the desired characteristics of materials necessary for practical applications. Studies of inorganic materials with non-centrosymmetric structure, such as quartz, potassium dihydrogen phosphate, barium  $\beta$ -borate and others, have revealed deficiencies such as a low NLO response, slow response times, production difficulties, low optical quality, the absence of extended  $\pi$ -electron delocalization, and inability to adjust the resulting structure. These deficiencies have been addressed and resolved using organic materials, which allowed the formation of centrosymmetric structures due to dipole–dipole interactions. In addition, the order of the dipoles in organic materials can be induced by an external field, “freezing” it with external temperature influences. In this case, one could obtain a response comparable to or even greater than that for inorganic crystals. However, these materials have

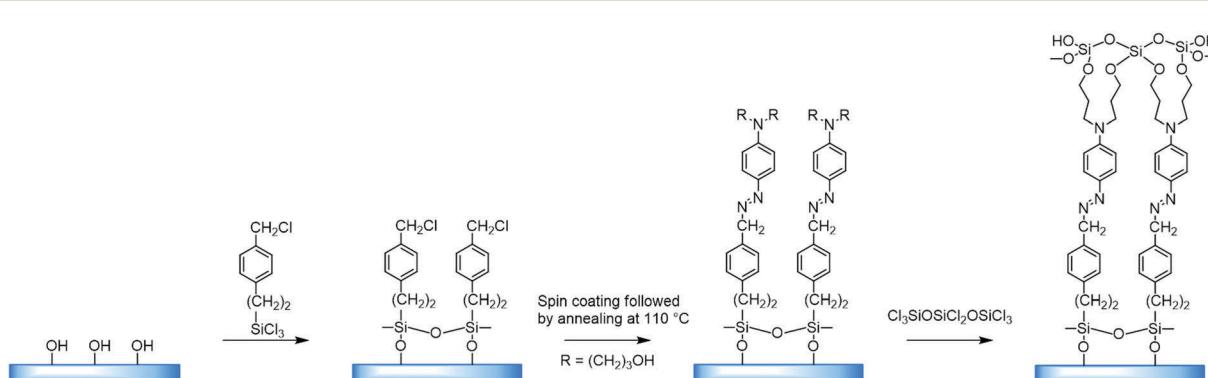


Fig. 7 Formation of a non-centrosymmetric multilayer of a film by combining self-assembly and a surface reaction of  $S_N2$  type. Adapted from ref. 28 (with permission from Copyright © 2016 American Chemical Society).

poor mechanical strength, poor stability and poor thermal stability, although they can be easily applied in the form of films for use in a device. Thus, in order to solve a number of significant problems with known materials, the main research focus today is aimed at the study of hybrid materials. The reason for this is the need to combine the advantages of organic and inorganic classes of compounds and level their shortcomings. Thus, one of the recent trends in non-linear optics is the investigation of the so-called metal-organic frameworks (MOFs) and coordination polymers (CPs), promising as future chemically derived metamaterials. Huge possibilities in terms of synthesis of new structures are reinforced with the variability of ions and clusters that can be used, allowing high precision and predictability in the properties of the final product.

## 2. Metal-organic frameworks

The last 15 years have been characterized by a great boom in the production and study of metal-organic frameworks and coordination polymers. These materials have proved to be excellent sorbents for gases, objects for storing hydrogen (as fuel), catalysts, elements of analytical devices for separating substances, as well as demonstrating unique physical properties in the field of ferroelectricity, non-linear optics, and magnetism. However, in addition to the existing broad scope of applications mainly due to a huge surface area, recently also unique features such as strong optical anisotropy in the visible region have been studied,<sup>32</sup> which were detected *via* reflection/absorption spectra (Fig. 8), depending on the direction of the polarization vector.

These properties hold great promise for use in the management of single crystals of MOFs in controlling photoexcitation under the influence of light, like non-linear and quantum metamaterials. For example, a highly stable metal-organic framework based on trinuclear iron(III) secondary building units (SBUs) linked by tetracarboxylic linkers with an anthracene core<sup>32</sup> may have unique metamaterial properties by ordering metal atoms (Fe chains) and dielectric elements in one crystal.

In general, MOFs are semiconducting or dielectric microcrystals whose nodes contain metal ions or polyvalent metal clusters connected by organic ligands (linkers). These may comprise both donor and acceptor groups, as well as forming primary structure elements which give rise to the manifestation of various physical properties. By combining the components and varying the synthesis parameters, one can produce a huge variety of structures for MOFs, thereby affecting the desired properties of the material. The size and shape of pores of the formed framework are determined by the length and functionality of the organic component.<sup>33</sup> Thus, by varying the initial molecular units, one can obtain final products with different architectures. One example is the use of heterocyclic molecules and molecules with hydrocarbon tails which allow for rotation around its axis, making it difficult to control the metal ion-ligand environment. In this case, preference is given to rigid aromatic carboxylate ligands, which possess high chelating

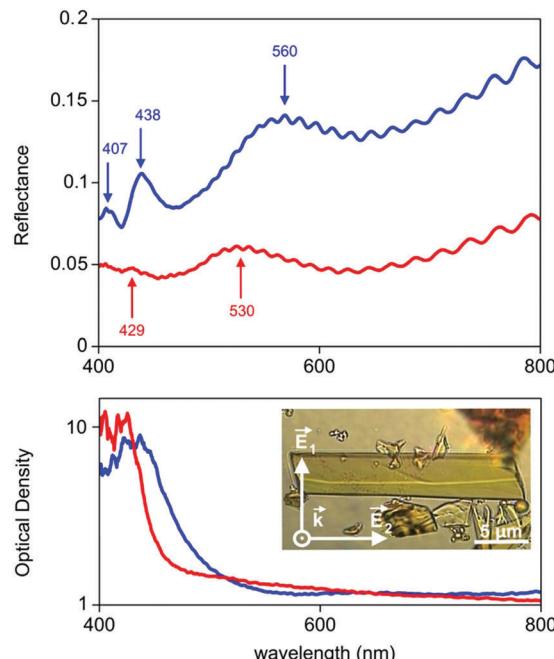


Fig. 8 Reflectance spectra and optical density of a single crystal of a MOF irradiated perpendicular to its surface. Red and blue curves correspond to reflection/absorption of electromagnetic waves with electric vectors  $E$  which are perpendicular ( $E_1$ ) or parallel ( $E_2$ ) to the longest side of the crystal, as shown in the inset. The slope of the optical density curves supports different band gaps of the MOF depending on the light polarization (2.7 eV for  $E_1$  and 2.4 eV for  $E_2$ ). Copyright Royal Society of Chemistry.

ability, allowing metal ions to be fixed in the nodes of the produced lattice.<sup>34</sup>

From a technical point of view, the major drawbacks of the existing methods for the preparation of MOFs are as follows: high energy costs, insufficient size of produced single crystals for studying their structural characteristics, low yield of the expected product, and labor-intensive methods for the synthesis and purification of the product. Equally important is instability in moist air and fragility, which severely limits the possibilities of practical applications. However, for the application of MOF crystals in metamaterials the main drawback is the lack of knowledge and methods for obtaining the order of structures in the area of more than 5 nm. The first priority questions arising from numerous studies are what parameters explicitly influence the produced framework architecture, how they can be corrected, and what relationships exist between the architecture and physical characteristics. In an attempt to determine the assumed relationships, a comparison of methods for the preparation of metal-organic frameworks with the organization of structures obtained during synthesis, as well as attempts to establish ways of developing non-linear optical properties of MOF crystals, is given below.

### 2.1 Measurements of non-linear optical properties of metal-organic frameworks

The recognized procedures to study the process of second harmonic generation are the Kurtz and Perry method (1968)



for powder samples, Hauchecorne (1971) for measurements in liquids, as well as for the measurement of individual nonlinearities for molecules in solution, Levine (1976), Oudar (1977), and finally, Jerphagnon and Kurtz (1970), who improved the procedure for the investigation of processes occurring in crystals, known as the Maker-fringe methods. Thus, successfully tested techniques for studying molecular, polycrystalline and single-crystal samples are universal for measuring response of various materials. The Kurtz and Perry method is the most frequently used for studying MOFs, despite the fact that powder analysis does not allow directed excitations along the crystal planes and thereby limits the analysis of properties in the area of metamaterials.

The method entails placing the sample in the form of a powder (about 50 mg) into a cell and irradiation with a low-energy laser (Fig. 9a). The intensity of second harmonic generation is measured using a photomultiplier. Results are usually compared with  $\alpha$ -quartz.

## 2.2 The relationship between the SHG and the size of the crystal

There is evidence that the SHG intensity depends on the crystal size. For materials such as  $\text{LiNbO}_3$ , where type 1 phase-matching, or index matching, occurs when the phase velocity of the fundamental radiation (1064 nm) equals the second harmonic (532 nm), the SHG intensity increases with the particle size and eventually reaches a plateau. In materials with no phase-matching (such as quartz), the intensity efficiency will reach a maximum value and then decrease, as the particle size increases. Therefore, the SHG measurement as a function of particle size is necessary for more accurate and reliable results.<sup>35</sup>

The efficiency of generation of second harmonic and other high-order nonlinearities (piezo effect, wave mixing, *etc.*) becomes essential with an increase in crystallinity of the sample. It is due to the fact that generation is directly related

to the type of crystal and its spatial orientation. This necessitates working in three dimensions with single crystals rather than powders and multicrystalline samples. We have, therefore, implemented the concept of optical studies of single crystals of MOFs (Fig. 9b). The concept is pretty simple, but very useful: a confocal system for detecting the optical signals (luminescence, reflection, second harmonic and others) is associated with two optical channels (or three, for luminescence). The optical channels can independently move in space due to the mechanical and optical elements in each channel; the single crystal is on a piezo stage which allows to locally excite it at any point and angle and collects the signals from anywhere. The excitation area is limited by diffraction but the collection area can be much smaller due to a confocal pinhole. The combination of different optical elements (Glan-Taylor prisms, quarter-wave plate, immersion objectives, *etc.*) in the channels allows to perform all optical experiments with the single crystal consecutively or two experiments simultaneously (*e.g.*, luminescence and transmission). This system is well described<sup>32</sup> and allows studying non-linear optical effects (exciton luminescence or birefringence) in different points of single microcrystals.

## 3. Metal–organic frameworks for non-linear optics

### 3.1 Producing metal–organic frameworks for non-linear optics

The question as to what method can successfully combine the possibilities of producing the desired crystal structure (inorganic component) and a variety of synthetic possibilities of the organic component has been posed by researchers for several years.

The main way to obtain the desired structure in the synthesis of metal–organic frameworks is the so-called strategy of secondary building units (SBUs).<sup>75</sup> This strategy is based on molecular or

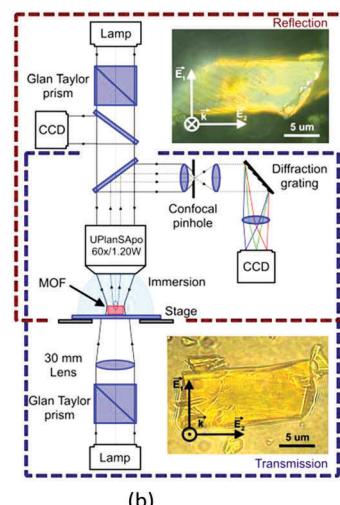
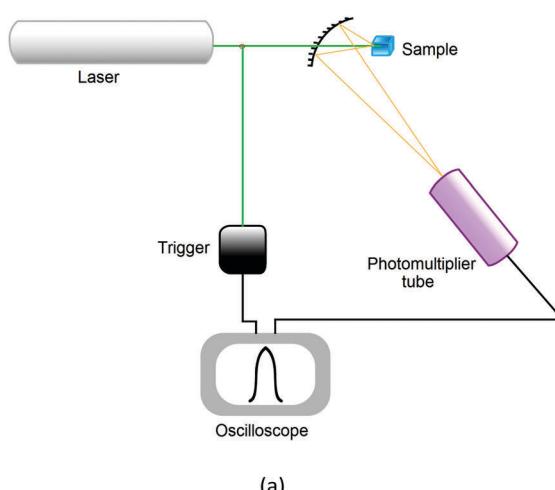


Fig. 9 (a) Schematic representation of the Kurtz method for powders. (b) Confocal multifunctional micro-spectroscopy setup for precision optical measurements with a single crystal of a MOF. The upper and lower channels are for reflectance and transmission measurements that can be upgraded to simultaneous luminescence measurements. Copyright Royal Society of Chemistry.



ionic assemblies and leads to the formation of secondary structural units. Non-centrosymmetric organization is a prerequisite for the generation of non-linear optical properties of bulk materials. For conventional methods of producing metal-organic frameworks, the creation of such non-centrosymmetric structures is challenging. The most successful SBUs may include metals connected with chiral ligands, since such materials were successfully used in selective catalysis and stereoselective separation.<sup>76</sup> Not only can the coordination of the organic ligand to a metal ion yield an increase in charge transfer transitions, but it also allows organic chromophores to assemble in highly ordered geometries like octahedra and tetrahedra, which is infrequently observed for organic materials. This interaction produces charge transfer in several directions. In addition, this is the type of structure that is relatively easy to obtain by selecting the right combination of metal ions and organic ligands. Table 1 summarizes various metal-organic frameworks and coordination polymers obtained by different approaches and their efficiency in second harmonic generation.

### 3.2 Ligands

**3.2.1 Multifunctional ligands.** One of the most successful methods for solving the problem of obtaining a non-centrosymmetric structure of a MOF was using a compound as a ligand which is capable of chelating a metal ion with several functional groups in different positions, *e.g.* deprotonated 2-hydroxy-nicotinic acid (nica).<sup>77</sup> 2-Hydroxy-nicotinic acid has several potential donor atoms which can allow high-dimensional structures to be obtained by removing one to three hydrogen atoms with careful adjustment of the pH of the medium, leading to various coordination modes. For example, investigations of second harmonic generation for crystals obtained by this method, by changing only the stoichiometry and pH, were reported.<sup>78</sup> This in turn leads to different degrees of ligand deprotonation, ultimately resulting in different coordination modes upon interaction with two different metal ions.

Deprotonated 2-hydroxy-nicotinic acid is capable of coordinating a metal using different functional groups, yielding four- and six-membered chelate rings (Fig. 10). The resulting compound  $\{[Zn_2(nica)_2(bpy)_{1.5}(H_2O)] \cdot 0.5(bpy) \cdot 3H_2O\}_n$  was obtained by hydrothermal synthesis and investigated by IR spectroscopy to determine possible coordination modes of the ligand with the metal ion. As a result, eight different coordination modes were suggested.

Single-crystal diffraction studies have shown that the resulting compound crystallized in the crystal class *mm2* (point group  $C_{2v}$ ),<sup>77</sup> in the orthorhombic non-centrosymmetric space group *Fdd2*. The N,O- and O,O-chelation of the  $Zn^{II}$  ions by nica<sup>2-</sup> results in the formation of zig-zag channels along the *c*-axis. The modest SHG response in the present case can be attributed partly to the partial cancellation of non-centrosymmetry between the intercrossing zig-zag chains as the dipole moments are cancelled out and partly to the absorption by the crystal in the presence of bpy guest molecules. Strong hydrogen interaction of the water molecules with the carboxyl group in nica<sup>2-</sup> occurs. Also, a strong  $CH \cdots \pi$  interaction (purple and blue dashed lines)

and a relatively weak  $\pi \cdots \pi$  interaction (green dashed line) are observed between the coordinated and guest bpy ligands. In the resulting non-centrosymmetric product, which crystallizes in the space group *Fdd2*, the polar axis lies along the zig-zag chain (Fig. 11). Polarity arises due to the location of the bridging nica<sup>2-</sup> ligands.

For this type of structure, non-linear optical properties were investigated in accordance with the methods proposed by Kurtz and Perry. The second harmonic generation intensity measured in a single crystal using a laser source is lower than that in  $SiO_2$ , and the corresponding peak is observed at 532 nm. The efficiency of second harmonic generation depends on several factors, such as non-centrosymmetry, intensity of push-pull dipole effects caused by ligands, donor-acceptor bonds in the framework, interaction between the guest molecules embedded in the structure and the functional groups of the ligands.<sup>78</sup> A small second harmonic response in this case can be explained by the partial absorption of the crystal due to the presence of guest bpy molecules, as well as by a decrease in non-centrosymmetry upon crossing the zig-zag chain because the dipole moments of the functional units are greatly reduced.

Another example is 2-(pyridin-4-yl)-1*H*-imidazole-4,5-dicarboxylic acid (H<sub>3</sub>pidc). In  $[Zn_2(pidc)(H_2O)Cl]_n$ , the asymmetric unit consists of two independent zinc ions, one completely deprotonated ligand, one coordinated water molecule and one coordinated  $Cl^-$  ion (Fig. 12).<sup>78</sup> The structure of the material in general is a two-dimensional grid, while hydrogen bonds allow the dimensionality of the system to be increased to 3D. A different coordination mode is observed in  $[Mn(Hpidc)(H_2O) \cdot 2H_2O]_n$ , in which three metal centers are chelated in an O,O- and N,O-bonding mode as well as a monodentate coordination of a pyridyl group (Fig. 13), eventually resulting in a three-dimensional network. Moreover, the considered structure also forms channels in the [001] direction filled with non-coordinating water molecules connected *via* hydrogen bonds. Various coordination modes allow the pyridine ring and imidazole to take up different positions. The fact that the carboxyl group and the imidazole ring cannot be located in the same plane ultimately leads to the formation of a non-centrosymmetric structure.<sup>78,79</sup> Coordination of the ligand increases its rigidity, resulting in a donor-acceptor system, which is necessary for the manifestation of a non-linear optical effect. The SHG measurement results showed that the magnitude of response in the first case was comparable with urea, while in the second case the value was about 0.8 relative to urea.<sup>78</sup> With regard to symmetry,  $[Zn_2(pidc)(H_2O)Cl]_n$  belongs to the class *222* and point group symmetry *D<sub>2</sub>*, while complex  $[Mn(Hpidc)(H_2O) \cdot 2H_2O]_n$  crystallizes in the space group *Cc*, which belongs to a polar point group *C<sub>s</sub>*.

Different structures were observed for compounds with *p*-pyridine carboxylato or *m*-pyridine carboxylato ligands (Fig. 14).<sup>80</sup> Although the difference is small, *m*-pyridine carboxylate led to the formation of 2D structures, in contrast to *p*-pyridine carboxylate, which gave 3D structures. This type of 2D structure is formed when a metal center is coordinated by unsymmetric bifunctional ligands.<sup>80</sup> If conditions are favorable, the 2D grids may form a non-centrosymmetric crystal.<sup>80</sup> In addition, *m*-pyridine





Table 1 Metal-organic frameworks (MOFs) and coordination polymers (CPs) with second harmonic generation (SHG) properties

Entry	Composition	Ligand	Space group	SHG	Ref.
1	[Zn <sub>2</sub> (X)(EOH) <sub>4</sub> ·3H <sub>2</sub> O] <sub>n</sub>	X = tetrakis[4-(carboxyphenyl)(oxamethyl)]methane acid	Cc	2.5 × KDP	36
2	[Cd(I <sub>n</sub> -N <sub>3</sub> ) <sub>2</sub> (H <sub>2</sub> O) <sub>2</sub> ] <sub>n</sub>	L = <i>trans</i> -2,3-dihydro-2-(4'-pyridyl)-3-(3'-tetrazolyl)phenyl]benzof[e]indole	Aba2	80 × urea	37
3	{(NH <sub>2</sub> Me) <sub>2</sub> }[Cd <sub>2</sub> (bta) <sub>4</sub> ] <sub>n</sub>	Hbta = biphenyl-2,2'-6,6'-tetracarboxylic acid	C2/c	0.8 × KDP	38
4	[Cd(bpy)(SA)] <sub>n</sub>	bphy = 1,2-bis(4'-pyridyl)hydrazine, H <sub>2</sub> SA = succinic acid	I2 <sub>1</sub> 3	0.1 × KDP	39
5	[Zn(cpt)(atz)] <sub>n</sub>	Hatz = 5-amino-1 <i>H</i> -tetrazol, Hcpt = 4-(4 <i>H</i> -1,2,4-triazol-4-yl)benzoic acid	P2 <sub>1</sub> 2 <sub>1</sub> 2 <sub>1</sub>	0.13 × KDP	40
6	[Bi <sub>2</sub> O <sub>2</sub> (pydc)] <sub>n</sub>	pydc = pyridine-2,5-dicarboxylate	P2 <sub>1</sub>	80 × SiO <sub>2</sub>	41
7	[Mg <sub>2</sub> (btc)(CH <sub>3</sub> COO)(C <sub>4</sub> H <sub>9</sub> NO) <sub>3</sub> ] <sub>n</sub>	H <sub>3</sub> btc = 1,3,5-benzene-tricarboxylic acid	P2 <sub>1</sub> 2 <sub>1</sub>	5 × KDP	42
8	{[K <sub>2</sub> Co <sub>2</sub> (L <sub>c</sub> ) <sub>2</sub> ](EOH) <sub>6</sub> Cl}] <sub>n</sub>	H <sub>2</sub> L <sub>c</sub> = 25,27-bis(hydroxy carbonylmethoxy)-26,28-dimethoxy- <i>p</i> -tert-butylcalix[4]arene	I2 <sub>1</sub> 3	~0.8 × urea	43
9	{[Rb <sub>3</sub> Co <sub>2</sub> (L <sub>c</sub> ) <sub>3</sub> ](EOH) <sub>6</sub> Cl}] <sub>n</sub>	H <sub>2</sub> L <sub>c</sub> = 25,27-bis(hydroxy carbonylmethoxy)-26,28-dimethoxy- <i>p</i> -tert-butylcalix[4]arene	I2 <sub>1</sub> 3	~1.0 × urea	43
10	{[N(C <sub>3</sub> H <sub>7</sub> ) <sub>4</sub> ][Ga <sub>3</sub> (btc) <sub>4</sub> ] <sub>n</sub>	H <sub>3</sub> btc = 1,3,5-benzene-tricarboxylic acid	R3c	15 × SiO <sub>2</sub>	44
11	[Cd(amp) <sub>2</sub> Br <sub>2</sub> ·H <sub>2</sub> O] <sub>n</sub>	amp = 2-amino-5-nitropyridine	Pm2	2.1 × KDP	45
12	{[Ce(t-tar)(CH <sub>2</sub> OHCH <sub>2</sub> OH)(H <sub>2</sub> O)]Cl} <sub>n</sub>	t-H <sub>2</sub> tar = 1-tartaric acid	P2 <sub>1</sub>	0.5 × urea	46
13	{[Ce(t-tar)(CH <sub>2</sub> OHCH <sub>2</sub> OH)(H <sub>2</sub> O)]Cl} <sub>n</sub>	d-H <sub>2</sub> tar = d-tartaric acid	P2 <sub>1</sub>	0.5 × urea	46
14	[Co(L)(bimb)·EtOH·H <sub>2</sub> O] <sub>n</sub>	EtOH = 1,2-aminobis(1 <i>H</i> -imidazol-1-yl-methyl)bisphenyl	P2 <sub>1</sub> 2 <sub>1</sub> 2 <sub>1</sub>	0.3 × urea	47
15	[Co(L)(bimb)·EtOH·H <sub>2</sub> O] <sub>n</sub>	bimb = 4,4-bis(1 <i>H</i> -imidazol-1-yl-methyl)bisphenyl	P2 <sub>1</sub> 2 <sub>1</sub> 2 <sub>1</sub>	0.3 × urea	47
16	[Co(L)(bimb)·EtOH·H <sub>2</sub> O] <sub>n</sub>	L = (R)-4-(1-carboxyethoxy)phenoxyl-3-chlorobenzoic acid,	P2 <sub>1</sub> 2 <sub>1</sub> 2 <sub>1</sub>	0.3 × urea	47
17	[Ni <sub>2</sub> (H <sub>2</sub> O)(L) <sub>2</sub> (bpy) <sub>2</sub> ·H <sub>2</sub> O] <sub>n</sub>	bpy = 4,4'-bis(1 <i>H</i> -imidazol-1-yl-methyl)bisphenyl	P2 <sub>1</sub>	0.7 × urea	48
18	[Cs(L)·H <sub>2</sub> O] <sub>n</sub>	HL = 1-malic acid	P2 <sub>1</sub>	4.24 × KDP	49
19	[Zn <sub>2</sub> (L) <sub>3</sub> (NO <sub>3</sub> ) <sub>2</sub> ·2NO <sub>3</sub> ·2CH <sub>3</sub> CN] <sub>n</sub>	L = (1 <i>R</i> ,2 <i>R</i> )-1,2-bis(4-(1,2,4-triazolyl))cyclohexane	P4 <sub>1</sub> 2 <sub>1</sub> 2	0.9 × KDP	50
20	[Ag <sup>+</sup> ·OH <sup>-</sup> ·2H <sub>2</sub> O] <sub>n</sub>	L = (1 <i>R</i> ,2 <i>R</i> )-1,2-bis(4-(1,2,4-triazolyl))cyclohexane	P3 <sub>1</sub> 2 <sub>1</sub>	0.1 × KDP	50
21	[Zn(tzb)] <sub>n</sub>	H <sub>2</sub> tzb = 1 <i>H</i> -tetrazolate-5-butyl acid	Pca2 <sub>1</sub>	0.5 × KDP	51
22	[Zn(dba)(Hpyim) <sub>2</sub> ] <sub>n</sub>	H <sub>2</sub> dba = 4,4'-methylene-dibenzene acid, Hpyim = 2-(2-pyridyl)imidazole	Cc	0.8 × urea	52
23	[Sr(Hsp)(H <sub>2</sub> O) <sub>3</sub> ·H <sub>2</sub> O] <sub>n</sub>	H <sub>3</sub> sp = 5-sulfoisophthalic acid	Am <sub>2</sub>	4 × KDP	53
24	[Zn(dcm) <sub>2</sub> ] <sub>n</sub>	H <sub>2</sub> dcm = 2,5-dicarboxy-1-methylpyridinium chloride	Fdd2	7 × KDP	54
25	[Zn(imal) <sub>2</sub> ] <sub>n</sub>	imal = 2-(1 <i>H</i> -imidazole-1-yl)acetate	Cc	0.5 × urea	55
26	{[Cu(PPh <sub>3</sub> ) <sub>2</sub> ](L)][ClO <sub>4</sub> ] <sub>2</sub> ] <sub>n</sub>	L = <i>N,N</i> -(2-pyridyl)(4-pyridylmethyl)amine	Cc	1.5 × urea	56
27	[Mn(Hpdc)(H <sub>2</sub> O) <sub>2</sub> ·2H <sub>2</sub> O] <sub>n</sub>	H <sub>3</sub> pidc = 2-(pyridin-4-yl)-1 <i>H</i> -imidazole-4,5-dicarboxylic acid	Pna2 <sub>1</sub>	1.5 × KDP	57
28	[Ag <sub>2</sub> (Hpidc)] <sub>n</sub>	H <sub>3</sub> pidc = 2-(pyridin-4-yl)-1 <i>H</i> -imidazole-4,5-dicarboxylic acid	Cc	0.9 × urea	58
29	[Cd((S)-ptrz) <sub>2</sub> ] <sub>n</sub>	(S)-Hptrz = 5-[ <i>(2S)</i> -pyrrolidine-2-yl]-1 <i>H</i> -tetrazole	P2 <sub>1</sub>	0.35 × KDP	59
30	[Zn((S)-ptrz) <sub>2</sub> ] <sub>n</sub>	(S)-Hptrz = 5-[ <i>(2S)</i> -pyrrolidine-2-yl]-1 <i>H</i> -tetrazole	P2 <sub>1</sub>	0.40 × KDP	59
31	[Zn <sub>2</sub> (L) <sub>2</sub> (4,4'-bipy) <sub>2</sub> ] <sub>n</sub>	H <sub>2</sub> L = <i>S</i> ( <i>-</i> )-1,10-binaphtho-2,2'-diacetic acid, 4,4'-bipy = 4,4'-bipyridine	P1	0.17 × KDP	60
32	[Zn(L)(bium-4) <sub>2</sub> ] <sub>n</sub>	H <sub>2</sub> L = <i>S</i> ( <i>-</i> )-1,10-binaphtho-2,2'-diacetic acid, bium-4 = 1,1'-(1,4-butenediy)bis(imidazole)	P4 <sub>3</sub>	0.08 × KDP	60
33	[Zn <sub>3</sub> (L) <sub>3</sub> (btb)(H <sub>2</sub> O) <sub>2</sub> ] <sub>n</sub>	H <sub>2</sub> L = <i>S</i> ( <i>-</i> )-1,10-binaphtho-2,2'-diacetic acid, btb = 1,6-bis(1,2,4-triazol-1-yl)hexane	P2 <sub>1</sub>	0.10 × KDP	60
34	[Cu(1,3-bdc)(opy)] <sub>n</sub>	1,3-H <sub>2</sub> bdc = isophthalic acid, opy = 4,4'-(oxybis(4,1-phenylene))dipyridine	P6 <sub>5</sub>	0.3 × urea	61
35	[LuCl <sub>3</sub> (H <sub>2</sub> L)(MeOH)·H <sub>2</sub> O] <sub>n</sub>	H <sub>2</sub> L = <i>N,N'</i> -bis(salicylidene)-1,2-cyclohexanediamine	Pna2 <sub>1</sub>	0.4 × urea	62
36	[Mg(L)(DMF)(H <sub>2</sub> O) <sub>3</sub> ] <sub>n</sub>	H <sub>2</sub> L = 4,4'- <i>(9,9</i> -dimethyl-9 <i>H</i> -fluorene-2,7-diyl)dibenzoc acid	P2 <sub>1</sub>	0.4 × urea	63
37	[Zn(L)(DMF)] <sub>n</sub>	H <sub>2</sub> L = 4,4'- <i>(9,9</i> -dimethyl-9 <i>H</i> -fluorene-2,7-diyl)dibenzoc acid	Aba2	0.8 × urea	63
38	[Cd <sub>2</sub> L(H <sub>2</sub> O) <sub>2</sub> ·solvent] <sub>n</sub>	H <sub>4</sub> L = tetrakis[4-(carboxyphenyl)oxamethyl]methane acid, solvent = DMA/H <sub>2</sub> O	Fdd2	3.6 × KDP	64
39	Zn(bip)(py) <sub>2</sub> ] <sub>n</sub>	H <sub>2</sub> bip = 5-bromoisophthalic acid, py = pyridine	P2 <sub>1</sub> 2 <sub>1</sub>	0.7 × urea	65
40	[Zn <sub>2</sub> Cl(btc)(ttimb)] <sub>n</sub>	ttimb = 1,3,5-tris(1-imidazol-1-ylmethyl)-2,4,6-trimethylbenzene, H <sub>3</sub> btc = 1,3,5-benzene-tricarboxylic acid	P2 <sub>1</sub> 2 <sub>1</sub>	0.7 × urea	66



Table 1 (continued)

Entry	Composition	Ligand	Space group	SHG	Ref.
41	$[\text{Zn}_3(\text{btc})_2(\text{titmb})\text{H}_2\text{O}]_n$	titmb = 1,3,5-tris(1-imidazol-1-ylmethyl)-2,4,6-trimethylbenzene, $\text{H}_3\text{btc} = 1,3,5$ -benzene-tricarboxylic acid	$P\bar{2}_1$	0.8 × urea	66
42	$[\text{Zn}(\text{paba})_2\text{H}_2\text{O}]_n$	Hpaba = <i>p</i> -aminobenzoic acid	$P\bar{2}_12_1$	1 × KDP	67
43	$[\text{Zn}(\text{tzf})_2]_n$	Htzf = 1 <i>H</i> -tetrazole-5-formic acid	$\bar{I}\bar{4}2d$	1.1 × KDP	68
44	$[\text{NaZn}(\text{acac})_2(\text{AcO})\text{EtOH}]_n$	acac = acetylacetone	$P\bar{2}_12_1$	15 × KDP	69
45	$[\text{Cd}_2(4,4'\text{-bipy})_2(\text{H}_2\text{O})_3(\text{SO}_4)_2\text{H}_2\text{O}]_n$	4,4'-bipy = 4,4'-bipyridine	$C2$	0.27 × KDP	70
46	$[\text{Ag}_4(\text{dob})_4(\text{BF}_4)_4]_n$	dob = 1,3-di(2-oxazolinyl)benzene	$\bar{C}\bar{c}$	0.3 × urea	71
47	$[\text{Zn}(\text{L})_2]_n$	HL = 1,3-dithiole-2-tetrazole	$\bar{I}\bar{4}2d$	0.7 × KDP	72
48	$[\{\text{Ca}(\text{L})_2(\text{H}_2\text{O})_2\}_2\text{Cl}_2]_n$	L = $\beta$ -D-fructose	$P\bar{2}_1$	0.466 × KDP	73
49	$[\{\text{Ca}(\text{L})_2(\text{H}_2\text{O})_2\}_2\text{Cl}_2\text{H}_2\text{O}]_n$	L = $\beta$ -D-fructose	$C2$	0.122 × KDP	73
50	$[\text{Zn}_2(1,3\text{-bdc})(\text{ppa})_2(\text{H}_2\text{O})]_n$	ppa = pipemidic acid, $\text{H}_2\text{bdc} = 1,3$ -benzenedicarboxylic acid	$C\bar{c}$	5.6 × KDP	74

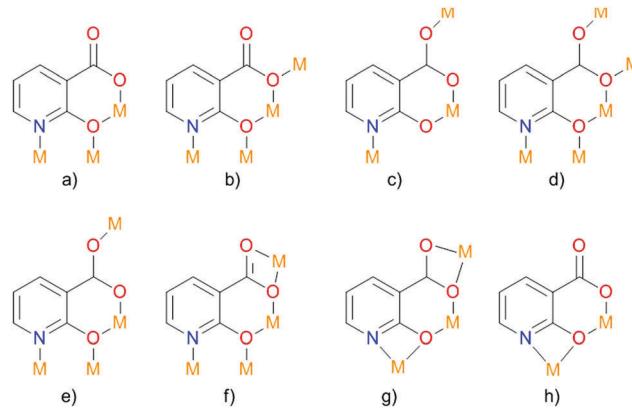


Fig. 10 Possible coordination modes of deprotonated 2-hydroxy-nicotinic acid. Adapted from ref. 77 with permission from © 1996–2016 MDPI AG.

carboxylates have their own electronic asymmetry, which also has a positive effect on the NLO properties.

Crystals of  $[\text{Cu}(\text{H}_2\text{Me}_4\text{BPZ})\text{Br} \cdot 0.5\text{H}_2\text{O}]_n$  ( $\text{H}_2\text{Me}_4\text{BPZ} = 3,3',5,5'$ -tetramethyl-4,4'-bipyrazole), obtained by the reaction of CuBr and  $\text{H}_2\text{Me}_4\text{BPZ}$  under solvothermal conditions, not only demonstrate a SHG value comparable to that of KDP,<sup>81</sup> but also exhibit ferroelectric properties. In this species, a 3D porous structure is formed by connecting four adjacent binding ligands of  $\text{H}_2\text{Me}_4\text{BPZ}$  (Fig. 15), which in turn are connected by serrated channels of  $[\text{CuBr}]$ .<sup>81</sup>

It is, however, important that the used compounds are stable under the influence of laser radiation, in contrast to, for example, coordination polymers,<sup>82</sup> so that studying the second harmonic generation is technically possible. For example, measurements of the second harmonic generation effect failed for the non-centrosymmetric thallium(i) coordination polymer  $[\text{Ti}_4(\text{adc})(\text{ox})]_n$  (adc = acetylenedicarboxylate, ox = oxalate) (Fig. 16) since decomposition by the laser beam occurred, releasing potentially toxic thallium(i) compounds.<sup>82</sup>

Previously, an attempt was made to study how the introduction of another metal in the metal–organic framework structure will affect the non-linear optical properties of the system. The first work on heterometallic compounds was focused on  $\text{Cd}^{2+}$  and  $\text{Li}^+$  ions. It was shown that the addition of alkali and alkaline earth metals to the framework could significantly enhance the SHG.<sup>83</sup> The first frameworks were synthesized in the presence of  $\text{LiNO}_3$  or  $\text{NH}_4\text{NO}_3$  under solvothermal conditions.<sup>83</sup> Substituted 1,3- or 1,4-benzenedicarboxylate were employed in the synthesis of heterometallic compounds,  $\{(\text{NH}_2\text{Me}_2)[\text{CdLi}(\text{p-bdc})_2] \cdot 3\text{CH}_3\text{OH}\}_n$  (1),  $\{(\text{NH}_2\text{Me}_2)_2[\text{Cd}(\text{p-bdc})_2]_3 \cdot (\text{NHMe}_2)_2\}_n$  (2),  $\{(\text{NH}_2\text{Me}_2)_2[\text{CdLi}(\text{m-bdc})_2]\}_n$  (3),  $\{(\text{NH}_2\text{Me}_2)_2[\text{Cd}(\text{OH-m-Hbdc})(\text{OH-m-bdc})_3] (\text{DMF})_3(\text{CH}_3\text{OH}) \cdot 3\text{H}_2\text{O}\}_n$  (4),  $[\text{Cd}_{1.5}\text{Li}(5\text{-}t\text{-Bu-m-bdc})_2(\text{DMF})_2]_n$  (5),  $\{(\text{NH}_2\text{Me}_2)_2[\text{CdLi}(\text{oba})_2]\}_n$  (6),  $\{(\text{NH}_2\text{Me}_2)_2[\text{CdLi}(\text{sdba})_2] \cdot 3\text{DMF} \cdot \text{CH}_3\text{OH}\}_n$  (7), and  $\{(\text{NH}_2\text{Me}_2)_2[\text{Cd}_3(\text{H}_2\text{O})_3(\text{OH})(\text{sdba})_3]\}_n$  (8) (*p*-H<sub>2</sub>bdc = 1,4-benzenedicarboxylic acid, *m*-H<sub>2</sub>bdc = 1,3-benzenedicarboxylic acid, OH-*m*-H<sub>2</sub>bdc = 5-hydroxy-1,3-benzenedicarboxylic acid, 5-*t*-Bu-*m*-H<sub>2</sub>bdc = *tert*-butyl-1,3-benzenedicarboxylic acid, H<sub>2</sub>oba = 4,4'-oxybis(benzoic acid) and H<sub>2</sub>sdba = 4,4'-sulfonyldibenzoic acid), using

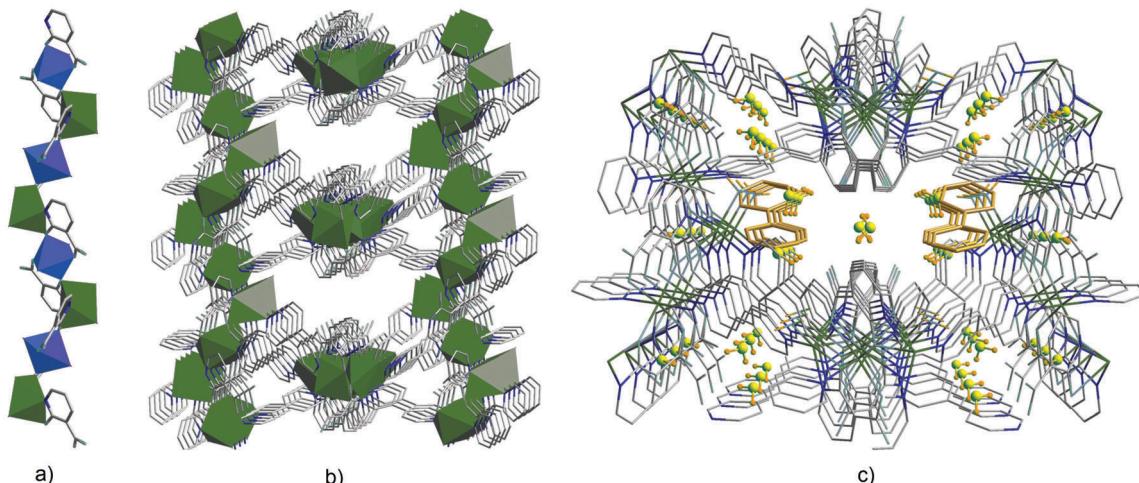


Fig. 11 (a) Formation of metal chains in  $\{[Zn_2(nica)_2(bpy)_{1.5}(H_2O)] \cdot 0.5(bpy) \cdot 3H_2O\}_n$  through coordination of  $nica^{2-}$  ( $H_2nica$  = nicotinic acid) ligands in a chelating manner; (b) the 3D framework in  $\{[Zn_2(nica)_2(bpy)_{1.5}(H_2O)] \cdot 0.5(bpy) \cdot 3H_2O\}_n$  viewed along the  $c$ -axis; (c) incorporation of guest molecules such as water and 4,4'-bipyridine. Adapted from ref. 77 with permission from © 1996–2016 MDPI AG.

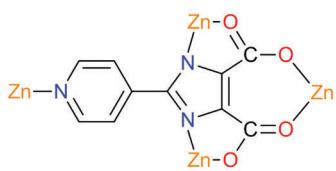


Fig. 12 Section of the structure of  $[Zn_2(picd)(H_2O)Cl]_n$ .<sup>78</sup>

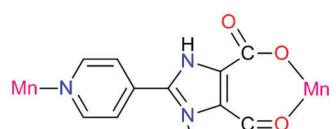


Fig. 13 Section of the structure of  $[Mn(Hpicd)(H_2O)_2H_2O]_n$ .<sup>78</sup>

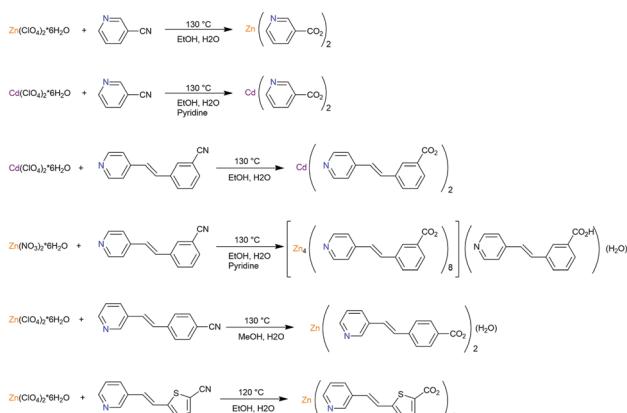


Fig. 14 Synthesis of 2D coordination polymers consisting of  $d^{10}$  metals and bridging  $p$ -pyridine carboxylato ligands. Adopted from ref. 80.

the solvothermal method.<sup>83</sup> Six compounds (2, 3, 4, 6, 7, and 8) were also investigated qualitatively based on the fact that they all crystallize in acentric (3) or chiral space groups (2, 4, 6, 7, and 8).

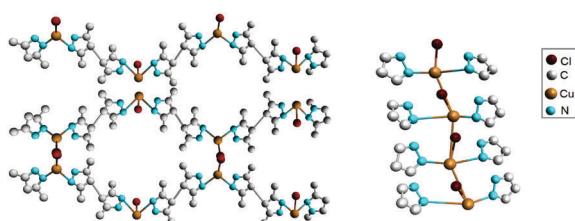


Fig. 15 Part of the crystalline structure of  $[Cu(H_2Me_4BPZ)Br \cdot 0.5H_2O]_n$ . Left: 1D serrated channels extending along the  $c$ -axis; right: side view of the 3D framework. Adapted from ref. 81. Copyright © 2016 Elsevier B.V.

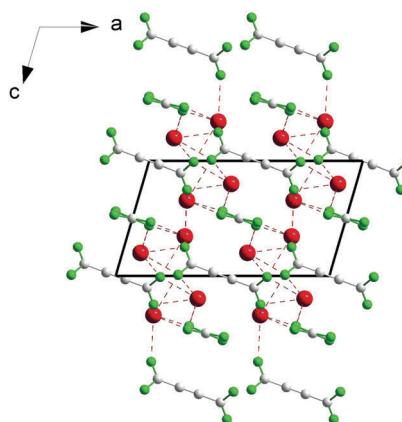


Fig. 16 Schematic diagram for  $[Tl_4(adc)(ox)]_n$  in the  $[010]$  projection. Adapted from ref. 82. Copyright © 2016 Elsevier.

Compound 2 exhibited the lowest SHG value, which is nonetheless still about two times greater than quartz, due to the fact that the shortest ligand with a lower degree of conjugation was used. Compound 6 exhibited the highest value, namely 800 times greater than quartz, and has good prospects, since its characteristics are better than that of technologically important  $LiNbO_3$  (600 times greater SHG than that of quartz).

The exceptions to the series are **1** and **5**, since they crystallize in centrosymmetric structures.

However, the use of the proposed strategy<sup>83</sup> was not successful in other cases; for example,  $[\text{CdCa}(m\text{-bdc})_2(\text{DMF})_2]_n$ ,  $[\text{CdCa}(\text{OH}\text{-}m\text{-bdc})_2(\text{H}_2\text{O})_2\text{NHMe}_2]_n$ , and  $[\text{CdSr}_2(\mu\text{-bdc})_2(\text{NO}_3)_2(\text{DMF})_4]_n$  show weak SHG values, despite crystallizing in non-centrosymmetric structures.<sup>84</sup>

**3.2.2 Chiral ligands.** It should first be noted that all chiral MOFs are active for SHG,<sup>85</sup> and there are some basic synthetic approaches to produce a chiral MOF. However, one should bear in mind that the presence of a chiral structure does not necessarily guarantee a high response of second harmonic generation. Perhaps this is one of the reasons why researchers are on an alternative path to produce non-centrosymmetric structures.

A metal-organic framework based on a trinickel secondary building unit was produced using a ligand with two chiral centers, which simultaneously serve as the donor and the acceptor in the system (Fig. 17). Such properties of a ligand are necessary for the generation of non-linear optical properties. The cyano group can be converted to a tetrazole in a click reaction with  $\text{NaN}_3$ . The molecule with two chiral centers (Fig. 17a), which has no symmetry, also crystallizes in a non-centrosymmetric space group. The significant enhancement of second harmonic generation was related to the cumulative effect caused by a high degree of asymmetry of the ligand due to the so-called pull-push effect and the metal-ligand coordination.<sup>86</sup>

Coordination polymers  $[\text{Zn}(\text{bindc})\text{Cl}_2]_n$ ,  $[\text{Zn}(\text{bindc})\text{Br}_2]_n$  and  $[\text{Ag}(\text{bindc})\text{NO}_3]_n$  ( $\text{bindc} = N,N'\text{-bis}(\text{isonicotinoyl})\text{-}(1R,2R)\text{-diaminocyclohexane}$ ) forming a helical superstructure, which is known to result in considerable enhancement of second harmonic generation, were constructed from the  $C_2$ -symmetrical chiral  $N,N'\text{-bis}(\text{isonicotinoyl})\text{-}(1R,2R)\text{-diaminocyclohexane}$  ( $\text{bindc}$ ) and  $\text{Zn}^{II}$ ,  $\text{Ag}^I$  and  $\text{Cu}^I$  *via* complementary hydrogen interactions of the bis(amido) ligands.<sup>87</sup> To ensure the formation of a highly dimensional structure, ligands containing amide groups were selected, because these groups provide strong hydrogen interactions with predictable regularity, which is particularly useful for self-assembly processes.<sup>87</sup>

Measurements using the Kurtz-Perry method yielded values of 0.8 U (1 U = SHG of urea) for bindc, even though this molecule does not have strong pull-push effects. This is due

to the helical superstructure in bindc. This is not observed in coordination polymers or MOFs, where the pyridyl nitrogen atom is involved in the coordination to metal ions.<sup>87</sup>

The zinc complex  $[\text{Zn}(\text{bindc})\text{Cl}_2]_n$  has a tetrahedral coordination, including two chloro ligands and two nitrogen atoms of two different bindc molecules. Coordinated dipyridyl of the ligand molecules forms an extended one-dimensional channel along the *b*-axis, complementary amide hydrogen bonds form one-dimensional zig-zag chains “parallel” to one another along the crystallographic *c*-axis, and the chloro ligands approach the pyridyl hydrogen atoms along the *a*-axis, resulting in a three-dimensional network. The replacement of zinc(II) with other metal(II) ions does not lead to significant changes in the second harmonic generation, but affects the thermal stability of the material.<sup>87</sup>

It follows from the above that non-centrosymmetric MOFs, which can be obtained by employing aromatic chiral ligands or asymmetric ligands with a pull-push effect (with an electron donor-acceptor system), can replace the existing materials for NLO, but the synthetic methods are complicated and chiral ligands can be expensive. An alternative synthetic strategy is so-called spontaneous resolution based on achiral ligands. Spontaneous resolution is the separation of racemates into their enantiomers in the form of conglomerates, which is a rare phenomenon. To obtain a non-centrosymmetric or chiral porous material by spontaneous resolution, it is preferable to use rigid achiral ligands with low symmetry. For example, deprotonated pyridine-3,5-dicarboxylic acid (3,5-H<sub>2</sub>PDC) was employed as a rigid V-shaped tridentate ligand with a lower symmetry ( $C_{2v}$ ) compared to typical tridentate ligands of deprotonated 1,3,5-benzene tricarboxylic acid ( $D_{3d}$  symmetry).<sup>88</sup> One advantage of dianionic 3,5-PDC in the coordination of a divalent metal is the formation of neutral compounds.<sup>88</sup>

It was shown that the solvent plays a key role in the preparation of metal-organic frameworks influencing the final structure. Using spontaneous resolution under solvothermal conditions and solvents such as DMF and ethylene glycol, a chiral MOF,  $[\text{Mn}(3,5\text{-PDC})(\text{H}_2\text{O})(\text{glycol})]_n$  with helical channels (Fig. 18), and two isostructural non-centrosymmetric MOFs,  $[\text{M}(3,5\text{-PDC})(\text{H}_2\text{O})(\text{glycol})\cdot\text{H}_2\text{O}]_n$  ( $\text{M} = \text{Co, Ni}$ ), were synthesized. Unfortunately, measuring second harmonic generation was outside the scope of this study.<sup>88</sup> However, this work demonstrates a new synthetic route for so-called spontaneous resolution, which turned out to be perfectly suited for the preparation of chiral and non-centrosymmetric MOFs.

It was also shown that even weak interactions between guest molecules and the framework can cause a transition from a chiral to an achiral structure depending on the size of the guest species. One example is  $[\text{Zn}(\text{ain})_2]$  ( $\text{ain} = 2\text{-aminoisonicotinic acid}$ ) in which a transition from a chiral to an achiral crystal was observed depending on the solvent used in the synthesis (DMA or DMF). The compounds  $[\text{Zn}(\text{ain})_2(\text{DMA})]_n$  and  $[\text{Zn}(\text{ain})_2(\text{DMF})]_n$  have the same structural network, but slightly different crystal symmetries, and their channels have a different shape,<sup>89</sup> due to the different sizes of DMA and DMF.<sup>89</sup> Such seemingly small differences can lead to a probable variation of the second

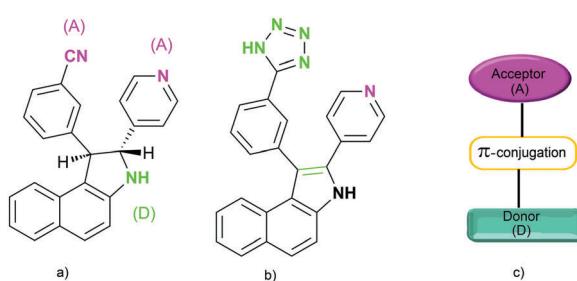


Fig. 17 (a) Ligand reported by Xiong *et al.*; (b) the corresponding *in situ* generated ligand; and (c) schematic illustration of a donor-acceptor system. Adapted from ref. 86 (Copyright © 2005, American Chemical Society).



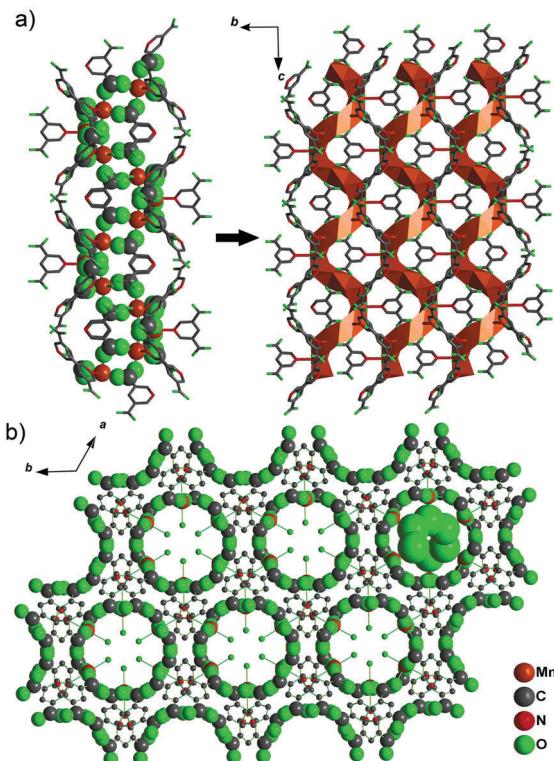


Fig. 18 (a) The Mn–O–C chain along the  $6_1$  axis connected to each other to form the chiral JUC-58 framework and (b) the network of JUC-58 looking down the  $6_1$  axis (001 direction), showing the homochiral water chain included in the channels. Adapted from ref. 88 (Copyright © 2013 Elsevier).

harmonic values. It is interesting to note that simply immersing the obtained product in another solvent does not result in structural changes.<sup>89</sup>

Another interesting case is a series of pillared-layer metal–organic frameworks  $[\text{Cd}(5\text{-aip})(\text{bpy})\cdot 1.5\text{DMA}]_n$ ,  $[\text{Cu}(5\text{-aip})(\text{bpy})\cdot 1.3\text{DMA}]_n$ ,  $[\text{Co}(5\text{-aip})(\text{bpy})\cdot 1.6\text{DMA}]_n$ , and  $[\text{Cd}(5\text{-aip})(\text{bpy})\cdot 0.5(\text{H}_2\text{O})\cdot 1.3\text{DMA}]_n$  derived from 5-aminoisophthalic acid (Haip), 4,4'-bipyridine (bpy) and  $\text{M}^{\text{II}}$ . A symmetry change from chiral to achiral occurs by soaking the crystals in methanol at room temperature. This fact is explained by a change in the torsion angle of the aromatic rings of the bipyridine ligands after substitution of DMA molecules in the channels with methanol.<sup>90</sup>

Strategies for potentially adjusting the symmetry of crystals have been reported.<sup>91</sup> However, one should always consider the solvent effect to control changes in lattice parameters with a change in crystal symmetry, which may affect the mechanical properties of the material. Also, substitution conditions must be chosen which ensure that the process proceeds to completion.

Also, apart from using asymmetric ligands, there is a strategy of using racemates. One example is diamond-like  $[\text{Cd}(\text{Imazethapyr})_2]_n$  which crystallizes in a non-centrosymmetric space group ( $Fdd2$ ) belonging to a polar point group ( $C_{2v}$ ). This metal–organic framework is obtained by the reaction of racemic H-Imazethapyr with  $\text{Cd}(\text{ClO}_4)_2\cdot 6\text{H}_2\text{O}$  under hydrothermal conditions (Fig. 19).<sup>92</sup> There are no network interpenetrations in the final structure due to steric constraints and short distances between two

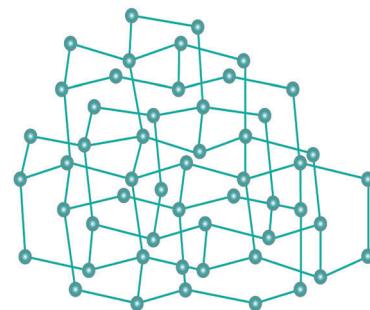


Fig. 19 A simplified 3D diamond-like structure of  $\text{Cd}(\text{Imazethapyr})_2$ . Straight lines resemble ligands, balls correspond to metal ions. Adapted from ref. 92 (Copyright © 2002, American Chemical Society).

adjacent cadmium ions. The second harmonic generation value for  $\text{Cd}(\text{Imazethapyr})_2$  is 20 times higher than that of potassium dihydrogen phosphate.

### 3.3 Diamondoid structures

The most likely candidates for producing crystals with non-linear optical properties are metal ions, which have a coordination number of 4 or 8 and are connected *via* a bifunctional ligand. Such components tend to form the so-called diamond structure (Fig. 20).<sup>89,91,93</sup> According to several studies, diamondoid structures are defined as promising in terms of obtaining non-centrosymmetric structures. Crystals of diamond-type inherently crystallize in a centrosymmetric space group ( $Fd\bar{3}m$ ), so that the inversion center is in the middle of the C–C bond between two adjacent nodes. To obtain non-centrosymmetric structures, non-centrosymmetric bridging ligands and different adjacent connecting nodes (Fig. 21) can be used.<sup>80</sup> Assuming a diamondoid topology with an Si–O–Si bond angle of  $144^\circ$ , one can produce V-shaped bonds between two adjacent Si atoms, which generally results in non-centrosymmetric structures (Fig. 22).

The diamondoid structure can be adjusted by changing the length of the ligand in the system. Thus, changing the length of the *p*-pyridine carboxylato ligand in  $\text{Zn}^{2+}$ - or  $\text{Cd}^{2+}$ -based two-dimensional coordination polymers resulted in non-centrosymmetric diamondoid structures which showed a

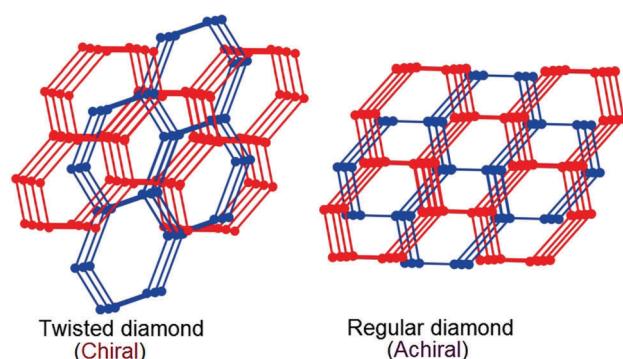


Fig. 20 Topological view of the structural transformation from the chiral dia net to the achiral dia net. Adapted from ref. 89 (© Royal Society of Chemistry 2016).



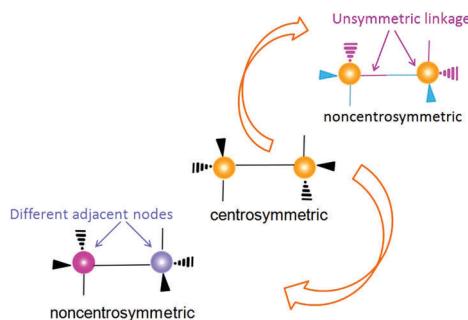


Fig. 21 Two methods for removing an inversion center in a diamondoid structure: (top) using unsymmetrical linkage and (below) using different adjacent nodes. Adapted from ref. 80 (Copyright © 2012, American Chemical Society).

higher SHG with a systematic increase in the length of the ligand,<sup>93–95</sup> *i.e.* with increasing conjugation between the donor and the acceptor, which in turn increases hyperpolarizability of the chromophores. Here, it is shown for the first time that non-linear optical properties can be changed by changing the length of the bridging conjugated ligands. For example, the obtained materials exhibit an SHG intensity of up to 400 times that of  $\alpha$ -quartz.<sup>92</sup>

Tetrazoles can act as bridging ligands for  $d^{10}$  metal ions<sup>96–98</sup> yielding a tetrahedral coordination, which enables a diamondoid structure to form  $1H$ -tetrazolate-5-butyrato coordination polymers,  $[\text{Zn}(\text{tzb})]_n$  and  $[\text{Zn}(3\text{-ptz})_2]_n$  ( $\text{H}_2\text{tzb} = 1H$ -tetrazolate-5-butric acid, 3-ptz = 5-(3-pyridyl)tetrazole). There are no interpenetrations in the network of the resulting compound,  $[\text{Zn}(3\text{-ptz})_2]_n$ , which was obtained by hydrothermal synthesis, due to the short distance between the metal ions and large substituents on the tetrazole.

### 3.4 Zero-dimensional objects

As was shown in several studies,<sup>99,100</sup> cations with octahedral coordination, distorted by second-order Jahn-Teller effects, such as high-spin  $d^4$  (*e.g.*,  $\text{CrF}_2$ ), low-spin  $d^7$  (*e.g.*,  $\text{NaNiO}_2$ ), and  $d^9$  (*e.g.*,  $\text{Cu}^{2+}$ ), are quite an important factor for non-centrosymmetry of the structure. In the case of octahedral  $d^9$  configuration, the unpaired electron may occupy either the  $d_{z^2}$  or  $d_{x^2-y^2}$  orbital of the  $e_g$  set. If it occupies the  $d_{z^2}$  orbital, most of the electron density will be concentrated between the metal and the two ligands on the  $z$ -axis. Thus, there will be greater electrostatic repulsion associated with these ligands than with the other four in the  $xy$  plane. This asymmetric

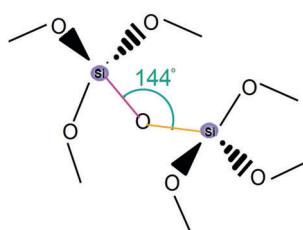


Fig. 22 An illustrative example for  $\alpha$ -quartz. Adapted from ref. 80 (Copyright © 2012, American Chemical Society).

distribution of the electron density may increase the overall energy of the system. Accordingly, elongation of bonds along the  $z$ -axis occurs, resulting in a lower symmetry.<sup>101–109</sup> Unfortunately, such an approach cannot warrant the expected result; therefore, the main task is controlling the macroscopic symmetry of the system.<sup>110</sup>

In the crystallization of a material, its symmetry may be affected not only by the selected cation, but also by its environment arising from the interaction of hydrogen bonds.<sup>110</sup> To determine the influence of hydrogen bonding on the orientation of the structure, two compounds were chosen: centrosymmetric  $[\text{NMe}_4]_2[\text{TiF}_6]$  and polar non-centrosymmetric  $[\text{C}(\text{NH}_2)_3]_2[\text{TiF}_6]$  (Fig. 23). With regards to non-linear optical properties, a value 25 times higher than that of  $\text{SiO}_2$  was obtained. The compounds under consideration have the same type of structure, the zero-dimensional  $\text{TiF}_6$  octahedra and organic cations, but they crystallize with different symmetries. In the case of  $[\text{C}(\text{NH}_2)_3]_2[\text{TiF}_6]$ , the alignment of the distorted octahedra is mainly responsible for the polar crystal structure and second harmonic generation, respectively. The eccentric distortion of the  $\text{TiF}_6$  octahedra is due to the interaction of hydrogen bonds between fluoride in the  $[\text{TiF}_6]^{2-}$  octahedra and the H-N bonds of the  $[\text{C}(\text{NH}_2)_3]^+$  cation, which are the key factor in determining the polar structure upon aligning the  $\text{TiF}_6$  octahedra. In contrast, interactions between the hydrogen

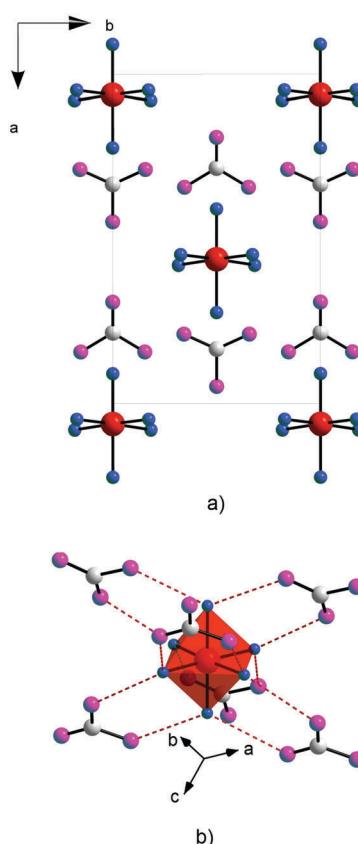


Fig. 23 Representation of the non-centrosymmetric structure of  $[\text{C}(\text{NH}_2)_3]_2[\text{TiF}_6]$ : (a) in the  $ab$  plane; and (b) showing the  $\text{F}\cdots\text{H}$  bonds. Adapted from ref. 110 (Copyright © 2011 Elsevier Inc.).



atoms in  $[\text{NMe}_4]^+$  and fluoride in  $[\text{TiF}_6]^{2-}$  in  $[\text{NMe}_4]_2[\text{TiF}_6]$  are extremely weak. The effect achieved in this case is incomparably smaller than strong hydrogen interactions present in  $[\text{C}(\text{NH}_2)_3]_2[\text{TiF}_6]$ .

### 3.5 Octupolar MOFs

Molecules possessing octupolar symmetry have long been the focus of investigations of non-linear optical properties. While dipoles tend to centrosymmetric organization, molecules with octupolar symmetry tend to form non-centrosymmetric structures.<sup>111</sup> It has been successfully demonstrated previously that a number of individual molecules with octupolar symmetry yield high second harmonic generation.<sup>111</sup> An example of a system with octupolar symmetry is an octupole consisting of charges in the corners of a cube, where positive charges are located at positions (1,0,0), (0,1,0), (0,0,1) and (1,1,1), and negative charges at points (0,0,0), (1,1,0), (1,0,1) and (0,1,1).<sup>111</sup> In organic chemistry, an example of an octupolar molecule is a benzene ring, in which electron donors and acceptors alternate (Fig. 24).<sup>111</sup> Such a molecule has high non-linear optical properties and does not have a permanent dipole moment, avoiding probable centrosymmetric crystallization.<sup>111</sup> Typical examples are molecules with a  $C_3$  symmetry axis, since it was shown that among the molecules with  $C_n$  symmetry axes, only molecules with a  $C_3$  axis do not have a dipole moment, but possess a finite hyperpolarizability component.<sup>111</sup>

With regards to MOFs, in order to obtain non-centrosymmetric octupolar structures with non-linear optical properties, trinuclear metal clusters can be used as building units. One example is  $\{[\text{Cd}_3(\mu_3\text{-OH})(\text{E}-4\text{-pyv-4-bza})_3(\text{py})_6](\text{ClO}_4)_2\}_n$  ( $\text{E}$ -4-pyv-4-bza = 4-[2-(4-pyridyl)ethenyl]benzoate) obtained from  $\text{Cd}(\text{ClO}_4)_2 \cdot 6\text{H}_2\text{O}$  and ethyl 4-[2-(4-pyridyl)ethenyl]benzoate using the hydro/solvothermal method.<sup>112</sup> Cadmium carboxylates, acting as secondary building blocks, give rise to a symmetry with 3-fold rotational symmetry (octupolar chromophores). The result is a chiral two-dimensional structure (Fig. 25). The measured signal of second harmonic generation has 130 times greater intensity than that of  $\alpha$ -quartz.<sup>112</sup>

Another example from this area is  $\{(\text{NH}_2\text{Me}_2)_2[\text{Cd}_3(\text{C}_2\text{O}_4)_4]\cdot\text{MeOH}\cdot 2\text{H}_2\text{O}\}_n$ , a MOF synthesized solvothermally from  $\text{CdCO}_3$  and oxalic acid.<sup>113</sup> This compound forms a three-dimensional porous network with a cubic acentric space group (Fig. 26). The unit in this case is two sets of equilateral triangles  $[\text{Cd}_3(\text{C}_2\text{O}_4)_4]^{2-}$  as a structural motif whose center lies on the threefold axis of rotation forming a hexameric  $[\text{Cd}_6(\text{C}_2\text{O}_4)_8]^{4-}$  ionic cluster.<sup>113</sup> Their interaction leads to an octupolar, 3D anionic supercage

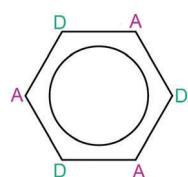


Fig. 24 An octupolar molecule with acceptors and donors alternating around a benzene ring.

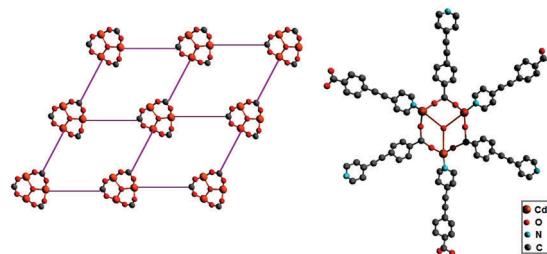


Fig. 25 Left: Schematic representation of the cationic building block for the octupolar structure of  $\{[\text{Cd}_3(\mu_3\text{-OH})(\text{E}-4\text{-pyv-4-bza})_3(\text{py})_6](\text{ClO}_4)_2\}_n$ . Right: Schematic representation of the 2D octupolar structure. Adapted from ref. 112. Copyright © 1999, American Chemical Society.

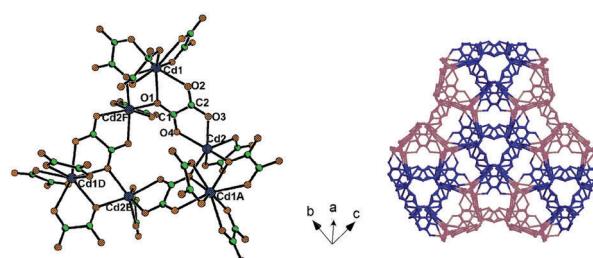


Fig. 26 Left: Representation of the 3D anionic network of  $\{(\text{NH}_2\text{Me}_2)_2[\text{Cd}_3(\text{C}_2\text{O}_4)_4]\cdot\text{MeOH}\cdot 2\text{H}_2\text{O}\}_n$ ; right: representation of the hexanuclear octupolar structure of  $[\text{Cd}_6(\text{C}_2\text{O}_4)_8]^{4-}$ . Adapted from ref. 113 (Copyright © 2007 John Wiley & Sons).

network with a rare, crystallographically imposed  $T_d$  symmetry. In such a system, there are small voids which are occupied by  $(\text{NH}_2\text{Me}_2)^+$  cations and solvent molecules (MeOH and  $\text{H}_2\text{O}$ ). The complete replacement of  $\text{NH}_2\text{Me}_2^+$  in  $\{(\text{NH}_2\text{Me}_2)_2[\text{Cd}_3(\text{C}_2\text{O}_4)_4]\cdot\text{MeOH}\cdot 2\text{H}_2\text{O}\}_n$  by  $\text{NH}_4^+$ ,  $\text{K}^+$ , or  $\text{Na}^+$  was accomplished and increases the resulting response to 155-, 110-, and 90-times that of  $\alpha$ -quartz, respectively. It appears that a decrease in the radius of the ion, which is exchanged, results in a decrease in the second harmonic generation response.<sup>113</sup>

### 3.6 Effect of the degree of interpenetration of grids

Second harmonic generation is strongly influenced by the degree of interpenetration of grids.<sup>114-117</sup> Interpenetration can occur when large voids are formed in a diamondoid crystal by extended ligands. Examples are the structures of  $[\text{Zn}(4\text{-pya})_n]$  and  $[\text{Cd}(4\text{-pya})_2\cdot\text{H}_2\text{O}]_n$ , [where 4-pya = (E)-3-(pyridin-4-yl)acrylate], which yield 5-fold interpenetration (Fig. 27) and crystallize in non-centrosymmetric space groups.<sup>114</sup> When using a ligand such as 4-(4-pyridyl)benzoate, the degree of interpenetration can be increased up to 7 (Fig. 28) in  $[\text{Cd}\{4\text{-}(4\text{-pyridyl})\text{benzoate}\}_2\cdot\text{H}_2\text{O}]_n$ ,<sup>117</sup> and for 4-[2-(4-pyridyl)ethenyl]benzoate, which contains an additional ethylene group, even up to 8 (Fig. 29) in  $[\text{Cd}\{4\text{-}(2\text{-}(4\text{-pyridyl})\text{ethenyl})\text{-benzoate}\}_2]_n$ .<sup>114</sup> Crystallization in a chiral space group is also possible in cases when the degree of interpenetration is even, for example 10.<sup>114,115</sup>

Unsymmetric ligands for achieving an unsymmetric distribution of electrons and eliminating the inversion center for MOFs with tetrahedrally coordinated metal centers (Fig. 30)



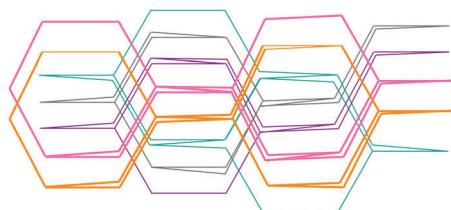


Fig. 27 Illustration of the maximum degree of interpenetration for degree 5. Adopted from ref. 114 with permission from © 1999–2016 John Wiley & Sons, Inc.

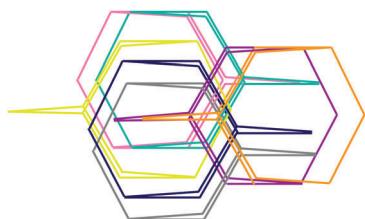


Fig. 28 Illustration of the maximum degree of interpenetration for degree 7. Adopted from ref. 117 with permission from © 2016 American Chemical Society.

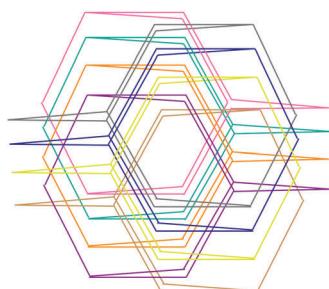


Fig. 29 Illustration of the maximum degree of interpenetration (for degree 8). Adopted from ref. 116 with permission from © Royal Society of Chemistry 2016.

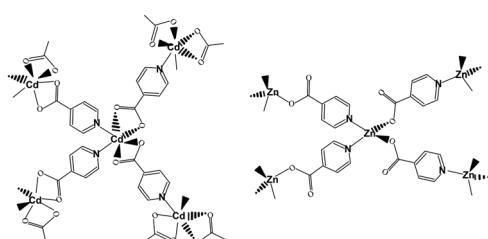


Fig. 30 3D MOFs of diamondoid topology built from tetrahedrally coordinated  $Zn^{2+}$  or  $Cd^{2+}$  and *p*-pyridine carboxylate linkers. Adopted from ref. 117. Copyright © 2016 American Chemical Society.

were first employed in 1999.<sup>117</sup> The degree of interpenetration may be significantly increased by increasing the length of the ligands, which promotes structures with more voluminous cavities. Since an extended ligand improves the interface between the donor and the acceptor, this in turn increases the molecular hyperpolarizability of the chromophores, which can improve the SHG of the crystal.<sup>8,11</sup>

### 3.7 1D coordination polymers: formation of channels

Obtaining non-centrosymmetric structures with 1D channels may be interesting even despite the fact that it is virtually impossible to control the structure in the other two dimensions. The first data on a 1D coordination polymer with non-linear optical properties were reported for cadmium chalcogenocyanates.<sup>118</sup> Anionic  $[Cd(XCN)_3]^-$  ( $X = S$  or  $Se$ ) polymeric chains may assume a non-centrosymmetric structure with all chalcogen atoms in the *trans* position to the nitrogen atoms (due to the *trans* effect) (Fig. 31). The formation of a non-centrosymmetric structure is strongly dependent on the symmetry of the cation. A compound with a crown-ether-metal cation with a center of inversion crystallizes in a centrosymmetric space group, while the monocation  $[K(18\text{-crown-6})]^+$ , which lacks a center of inversion, leads to the formation of a non-centrosymmetric material. The observed second harmonic generation in  $[K(18\text{-crown-6})][Cd(XCN)_3]$  is 15 times higher than that of quartz.

It is, however, also necessary to take into account the mutual orientation of the channels with respect to each other. For example, the compound  $[Co(pydc)(H_2O)_2]_n \cdot nH_2O$  ( $pydc$  = pyridine-3,4-dicarboxylate) yields only a subtle response due to the fact that the channels are oriented antiparallel to each other, which nullifies the dipole moment, thus preventing possible second harmonic generation.<sup>119</sup>

Creating a structure with some self-blockage may be of interest as well. In  $[Zn(pydc)]_n$ , three  $Zn^{2+}$  cations are connected by 4-[2-(4-pyridyl)ethenyl]cinnamate ligands, resulting in the formation of zig-zag channels. Parallel channels are connected by  $\pi$ - $\pi$  interactions.<sup>120</sup>

## 4. New applications of MOFs in non-linear optics: prospects

### 4.1 Third-order NLO

A new trend is the study of non-linear optical properties of third order for MOFs, which in turn has promise for use in all-optical switching in waveguides.<sup>121</sup> For example, crystals of  $[Cu_3(L)_2 \cdot 3nH_2O]_n$  and  $[CuAg_2(HL)_2]_n$  ( $H_3L$  = chelidamic acid) have the values  $1.67 \times 10^{-33}$  and  $1.52 \times 10^{-30}$  ESU,<sup>122</sup> which are comparable

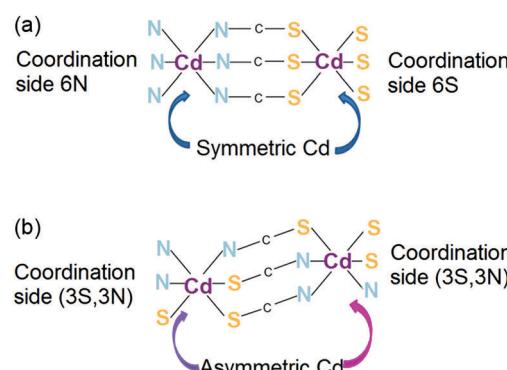


Fig. 31 Schematic representation of (a) centrosymmetric and (b) non-centrosymmetric channels of  $\{Cd(SCN)_3\}_n$ . Adapted from ref. 118. Copyright © 2016 Elsevier B.V.



to or even greater than the traditionally used crystals.<sup>123</sup> Thus, MOFs reveal themselves as promising materials in this field of third order non-linear effects.

## 4.2 Control of excitons

The crystalline and organic nature of MOFs allows the existence of Coulomb-bound electron-hole pairs (excitons). The potential offered by manipulating excitons when exposed to light and electric field is the basis for modern exciton transistors and polariton lasers. Semiconductors or special two-dimensional structures were typically used in such devices. Despite the fact that extremely fast and highly efficient manipulation of excitons can be obtained for such materials, there are limitations in the operating temperatures and device architectures. Given these shortcomings, an alternative is the use of ordered organic semiconductor structures which promote a shift of the operating temperature, additionally reducing the rate of exciton recombination. With regards to the problem of a short life cycle the solution is to use layered anisotropic materials. Taking into account the facts outlined above, it can be concluded that the combination of organic nature and crystal anisotropy is an effective solution for studying exciton control at ambient temperatures.

As proof-of-principle, a layered metal-organic framework  $[(\text{Zn}_2(\text{tbapy})(\text{H}_2\text{O})_2 \cdot 3.5\text{DEF})_n]$  (tbapy = 1,3,6,8-tetrakis (*p*-benzoate)-pyrene) was studied by Vinogradov, Hey-Hawkins *et al.*<sup>124</sup> The obtained adsorption, reflection and photoluminescence spectra indicate the presence of at least two types of excitons (Fig. 32). Since the compound has a layered structure, which limits the exciton motion in the same direction, they are divided into intra-layer, electrons and holes existing in the same layer, and inter-layer ones, located in different layers, respectively.

The study also demonstrated the efficient and independent control of excitons *via* disordering the structure or photoexcitation caused by light, which allows for achieving significant changes in the optical properties of single crystals. The said changes can either exist for a very short time or persist for several days. Thus, this material is promising for use in various excitonic devices.

## 5. Conclusions and outlook

In conclusion, in the production of NLO materials and even non-linear and quantum metamaterials, special attention should be paid to the class of metal-organic frameworks that combine the properties of organic and inorganic constituents. Particular attention should be paid to the selection of the metal-ligand pair. In order to further improve the optical properties of the produced materials, different parameters such as the length of the ligand, the degree of its substitution, the conjugation in the system, and the use of guest molecules with different natures should be adjusted. It should, however, be noted that the parameters of an obtained MOF crystal can be significantly affected by minor external influences, such as replacing solvent molecules with a different type of guest molecule, which in turn may lead to a change in the symmetry of the system. One should additionally take into account the fact that the best control in terms of prediction is provided for 3D structures, while 2D or 1D structures cannot be subjected to similar control of their architectures. It is also possible to change the degree of interpenetration of networks, which greatly affects the non-linear optical properties of the material.

While many efforts are being made to design MOF architectures that exhibit the desired properties, the possibility of studying the material structure at the molecular level and the measurement of its non-linear optical characteristics without destruction of the material remains an area with fairly significant problems.

The most important goal, however, is finding a correlation between the method of production, the synthesis parameters, the properties of the starting materials, and the resulting physical and chemical characteristics. Studies in this fundamental area can bear fruit for applied use as well. If MOFs are considered as a potential replacement for existing materials for practical applications in the field of non-linear optics, then thermal stability and mechanical strength are important. It is noteworthy that MOFs, which are able to compete in SHG and at the same time have acceptable thermal stability, already exist. This demonstrates that competent elaboration of these materials could improve their competitiveness with respect to technically important ones not only in terms of the desired non-linear optical effects, but also in the relative simplicity of preparation.

The development of metal-organic frameworks may transform into a very promising area, namely, obtaining metamaterials using chemical methods. A transition from physical methods of production to chemical ones would allow simplification of the process for obtaining specific materials, as well as

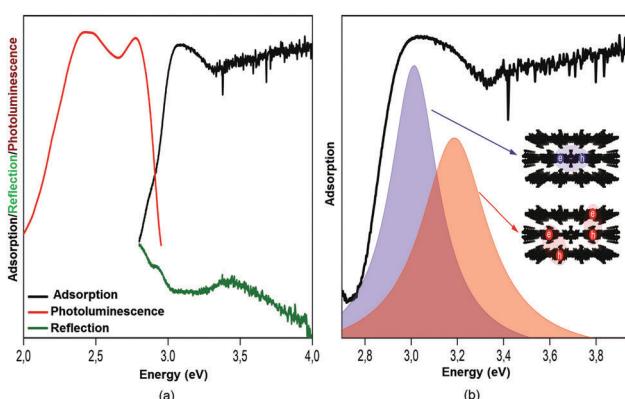


Fig. 32 The manifestation of excitons in optical spectra. (a) Absorption, reflection and photoluminescence spectra of single crystals of  $[(\text{Zn}_2(\text{tbapy})(\text{H}_2\text{O})_2 \cdot 3.5\text{DEF})_n]$ , placed on a quartz substrate in air. A single crystal is excited by continuous white light ( $100 \text{ mW cm}^{-2}$ ) to obtain absorption and reflection spectra, and by pulsed UV light (6 ps, 60 MHz, 3.1 eV,  $28 \text{ W cm}^{-2}$ ) to generate the photoluminescence. Peak/valley in absorption/reflection spectra near the fundamental absorption (3.4 eV) is addressed to exciton states. Photoluminescence peaks at 2.77 eV and 2.46 eV are coming from exciton states and organic components of  $[(\text{Zn}_2(\text{tbapy})(\text{H}_2\text{O})_2 \cdot 3.5\text{DEF})_n]$ , respectively. (b) Fit for the edge of absorption by intra- and inter-layer excitons with Lorentz distribution.



diversification of their final architecture. Attempts to produce metamaterials *via* chemical synthesis have been made, but it appears that this is not at all simple. However, more recently, it was possible to demonstrate that potential metamaterials can be obtained *via* chemical methods. In this respect, the development of MOFs which possess the desired properties characteristic of metamaterials *via* chemical methods is very promising.

## List of abbreviations

adc	Acetylenedicarboxylate	ima	2-(1H-Imidazole-1-yl)acetate
anp	2-Amino-5-nitropyridine	KDP	Potassium dihydrogen phosphate, $\text{KH}_2\text{PO}_4$
BBO	Barium borate, $\beta\text{-BaB}_2\text{O}_4$	LB	Langmuir–Blodgett
4,4'-bipy	4,4'-Bipyridine	MOF	Metal–organic framework
bimb	4,4-Bis(1H-imidazol-1-yl-methyl)biphenyl	NLO	Non-linear optics
bindc	<i>N,N'</i> -Bis(isonicotinoyl)-(1 <i>R</i> ,2 <i>R</i> )-diaminocyclohexane	opy	4,4'-(oxybis(4,1-phenylene))dipyridine
biim-4	1,1'-(1,4-Butanediyl)bis(imidazole)	ox	Oxalate
bpp	1,3-Di(pyridin-4-yl)propane	ppa	Pipemidic acid
bphy	1,2-Bis(4-pyridyl)hydrazine	<i>p</i> -na	<i>p</i> -Nitroaniline
btb	1,6-Bis(1,2,4-triazol-1-yl)hexane	py	Pyridine
CP	Coordination polymers	pyac	3-(4-Pyridyl)pentane-2,4-dionato
dcmp	2,5-Dicarboxy-1-methylpyridinium	pydc	Pyridine-2,5-dicarboxylate
dob	1,3-Di(2-oxazolinyl)benzene	(E)-4-pyv-4-bza	(E)-4-[2-(4-Pyridyl)ethenyl]benzoate
Hacac	Acetyl acetone	SBU	Secondary building unit
Hain	2-Aminoisonicotinic acid	SHG	Second harmonic generation
Haip	5-Aminoisophthalic acid	tbapy	1,3,6,8-Tetrakis( <i>p</i> -benzoate)pyrene
Hatz	5-Amino-1 <i>H</i> -tetrazol	tcf	2-Dicyanomethylene-3-cyano-4,5,5-dimethyl-2,5-dihydrofuran
Hbta	Biphenyl-2,2',6,6'-tetracarboxylic acid	titmb	1,3,5-Tris(1-imidazol-1-ylmethyl)-2,4,6-trimethylbenzene
Hept	4-(4H-1,2,4-Triazol-4-yl)benzoic acid		
Hpaba	<i>p</i> -Aminobenzoic acid		
(S)-Hptz	5-[(2 <i>S</i> )-Pyrrolidine-2-yl]-1 <i>H</i> -tetrazole		
Hpyim	2-(2-Pyridyl)imidazole		
Htzf	1 <i>H</i> -Tetrazole-5-formic acid		
<i>p</i> -H <sub>2</sub> bdc	1,4-Benzenedicarboxylic acid		
<i>m</i> -H <sub>2</sub> bdc	1,3-Benzenedicarboxylic acid, isophthalic acid		
OH- <i>m</i> -H <sub>2</sub> bdc	5-Hydroxy-1,3-benzenedicarboxylic acid		
5- <i>t</i> Bu- <i>m</i> -H <sub>2</sub> bdc	<i>tert</i> -Butyl-1,3-benzenedicarboxylic acid		
H <sub>2</sub> oba	4,4'-Oxybis(benzoic acid)		
H <sub>2</sub> bipa	5-Bromoisophthalic acid		
H <sub>2</sub> dba	4,4'-Methylenedibenzoic acid		
H <sub>2</sub> Me <sub>4</sub> BPZ	3,3',5,5'-tetramethyl-4,4'-bipyrazole		
H <sub>2</sub> nica	Nicotinic acid		
3,5-H <sub>2</sub> PDC	Pyridine-3,5-dicarboxylic acid		
H <sub>2</sub> SA	Succinic acid		
H <sub>2</sub> sdba	4,4'-Sulfonyldibenzoic acid		
ISAM	Ionic self-assembled monolayer		
L-H <sub>2</sub> tart	L-Tartaric acid		
D-H <sub>2</sub> tart	D-Tartaric acid		
H <sub>2</sub> tzb	1 <i>H</i> -Tetrazolate-5-butyric acid		
H <sub>3</sub> btc	1,3,5-Benzene-tricarboxylic acid		
H <sub>3</sub> pidc	2-(Pyridin-4-yl)-1 <i>H</i> -imidazole-4,5-dicarboxylic acid		
H <sub>3</sub> sip	5-Sulfoisophthalic acid		

## Acknowledgements

The authors are grateful to David Avnir for comprehensive support of the SCAMT lab and scientific advices. We gratefully acknowledge financial support from Leipzig University, the German Academic Exchange Service (DAAD, postdoctoral grant for A. V. V.), the ERASMUS + mobility program (A. V. V., L. M., E. H.-H.), the Russian Government Ministry of Education (this research was made possible due to funds provided aiming at maximising ITMO University's competitive advantage among world's leading educational centres), the Russian Foundation for Basic Research (grant no. 16-37-60073 mol\_a\_dk) and the EU COST Action CM1302 *Smart Inorganic Polymers* (SIPs). The authors thank Mr. Evgeny Uslamin (Eindhoven University of Technology) and Mr. Aleksandr V. Yakovlev (ITMO University, Saint Petersburg) for their help with the preparation of graphical material.

## References

- 1 B. E. A. Saleh and M. C. Teich, *Fundamentals of Photonics*, John Wiley & Sons, Inc, 2nd edn, 2007.
- 2 A. A. Blistanov, *Crystals of Quantum and Nonlinear Optics*, MISIS, Moscow, 2000, (in Russian).
- 3 B. N. Grechushnikov, I. S. Zheludev, A. V. Zalesski, S. A. Pikin, S. A. Semiletov, A. A. Urusovskaya, I. G. Chistyakov and L. A. Shuvalov, *Modern Crystallography. Physical Properties of Crystals*, Moscow, 1981, vol. 4 (in Russian).
- 4 C. Chen, Y. Wu and R. Li, *J. Cryst. Growth*, 1990, **99**, 790–798.
- 5 C. Chen, *J. Synth. Cryst.*, 2001, **30**, 36–42.
- 6 R. H. French, J. W. Ling, F. S. Ohuchi and C. T. Chen, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1991, **44**, 8496–8502.
- 7 W.-D. Cheng and J.-X. Lu, *Chin. J. Struct. Chem.*, 1997, **16**, 81–90.



8 D. Elmeri, L. Davis, S. Velsko, E. K. Graham and A. Zalkin, *J. Appl. Phys.*, 1987, **62**, 1968–1983.

9 L. K. Cheng, W. Bosenberg and C. L. Tang, *Prog. Cryst. Growth Charact.*, 1990, **20**, 9–57.

10 P. P. Fedorov, A. E. Kokh and N. G. Kononova, *Russ. Chem. Rev.*, 2002, **71**, 651–671.

11 F. Quist, C. M. L. Vande Velde, D. Didier, A. Teshome, I. Asselberghs, K. Clays and S. Sergeyev, *Dyes Pigm.*, 2009, **81**, 203–210.

12 L. Cheng, R. Foss, G. Meredith, W. Tam and F. Zumsteg, *Mater. Res. Soc. Symp. Proc.*, 1992, **247**, 27–39.

13 R. Jeng, Y. Chen, B. Mandal, J. Kumar and S. Tripathy, *Mater. Res. Soc. Symp. Proc.*, 1992, **247**, 111–117.

14 J. Wu, J. Valley, S. Ermer, E. Binkley, J. Kenney, G. Lipscomb and R. Lytel, *Appl. Phys. Lett.*, 1991, **58**, 225–227.

15 J. Wu, E. Binkley, J. Kenney, R. Lytel and A. Garito, *J. Appl. Phys.*, 1991, **69**, 7366–7368.

16 M. Era, T. Tsutsui and S. Saito, *Stud. Interface Sci.*, 1996, **4**, 287–321.

17 S. J. Lalama and A. F. Garito, *Phys. Rev. A: At., Mol., Opt. Phys.*, 1979, **20**, 1179–1194.

18 P. Guyot-Sionnest, W. Chen and Y. R. Shen, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1986, **33**, 8254–8263.

19 P. Ye and Y. R. Shen, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1983, **28**, 4288–4294.

20 T. F. Heinz, H. W. K. Tom and Y. R. Shen, *Phys. Rev. A: At., Mol., Opt. Phys.*, 1983, **28**, 1883–1885.

21 P. Guyot-Sionnest, W. Chen and Y. R. Shen, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1986, **33**, 8254–8263.

22 P. Guyot-Sionnest and Y. R. Shen, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1987, **35**, 4420–4426.

23 T. F. Heinz, C. K. Chen, D. Ricard and Y. R. Shen, *Phys. Rev. Lett.*, 1982, **48**, 478–481.

24 I. R. Girling, N. A. Cade, P. V. Kolinsky and C. M. Montgomery, *Electron. Lett.*, 1985, **21**, 169–170.

25 D. B. Neal, M. C. Petty, G. G. Roberts, M. M. Ahmad, W. J. Feast, I. R. Girling, N. A. Cade, P. V. Kolinsky and I. R. Peterson, *Electron. Lett.*, 1986, **22**, 460–481.

26 W. Lin, S. Yitzchaik, W. Lin, A. Malik, M. K. Durbin, A. G. Richter, G. K. Wong, P. Dutta and T. Marks, *Angew. Chem.*, 1995, **34**, 1497–1499.

27 X. Yang, D. McBranch, B. Swanson and D. Li, *Mater. Res. Soc. Symp. Proc.*, 1995, **392**, 27–32.

28 A. Ulman, *Chem. Rev.*, 1996, **96**, 1533–1554.

29 Z. Xu, T. Zhang, W. Lin and G. Wong, *J. Phys. Chem.*, 1993, **97**, 6958–6960.

30 G. Decher and J. D. Hong, *Thin Solid Films*, 1992, **210/211**, 831–835.

31 G. Decher, Y. Lvov and J. Schmitt, *Thin Solid Films*, 1994, **244**, 772–777.

32 A. V. Vinogradov, V. A. Milichko, H. Zaake-Hertling, A. Aleksovska, S. Gruschinski, S. Schmorl, B. Kersting, E. M. Zolnhofer, J. Sutter, K. Meyer, P. Lönncke and E. Hey-Hawkins, *Dalton Trans.*, 2016, **45**, 7183–7194.

33 J. L. C. Rowsell and O. M. Yaghi, *Microporous Mesoporous Mater.*, 2004, **73**, 3–14.

34 N. L. Rosi, M. Eddaoudi, J. Kim, M. O'Keefe and O. M. Yaghi, *CrystEngComm*, 2002, **68**, 401–404.

35 K. M. Ok, E. Ok Chia and P. S. Halasyamani, *Chem. Soc. Rev.*, 2006, **35**, 710–717.

36 Z. Guo, R. Cao, X. Wang, H. Li, W. Yuan, G. Wang, H. Wu and J. Li, *J. Am. Chem. Soc.*, 2009, **131**, 6894–6895.

37 Q. Ye, Y.-Z. Tang, X.-S. Wang and R.-G. Xiong, *Dalton Trans.*, 2005, 1570–1573.

38 L. Cheng, H. Hu, L. Zhang and S. Gou, *Inorg. Chem. Commun.*, 2012, **15**, 202–207.

39 Y. Xiao, J.-S. Guo, F.-K. Zheng and G.-C. Guo, *Inorg. Chem. Commun.*, 2016, **70**, 7–9.

40 L. Wang, Y. Ye, L. Zhang, Q. Chen, X. Ma, Z. Zhang and S. Xiang, *Inorg. Chem. Commun.*, 2015, **60**, 19–22.

41 A. C. Wibowo, M. D. Smith, J. Yeon, P. S. Halasyamani and H.-C. zur Loye, *J. Solid State Chem.*, 2012, **195**, 94–100.

42 S. Ma, J. A. Fillinger, M. W. Ambrogio, J.-L. Zuo and H.-C. Zhou, *Inorg. Chem. Commun.*, 2007, **10**, 220–222.

43 Y.-J. Liu, J.-S. Huang, S. Sin-Yin Chui, C.-H. Li, J.-L. Zuo, N. Zhu and C.-M. Che, *Inorg. Chem.*, 2008, **47**, 11514–11518.

44 D. W. Lee, V. Jo and K. M. Ok, *J. Solid State Chem.*, 2012, **194**, 369–374.

45 W.-W. Zhou, W. Zhao, B. Wei, F.-W. Wang, Y.-H. Chen, W.-Y. Fang and X. Zhao, *Inorg. Chim. Acta*, 2012, **386**, 17–21.

46 G. Hu, H. Zhang, H. Miao, J. Wang and Y. Xu, *J. Solid State Chem.*, 2015, **229**, 208–212.

47 X. Xu, Y.-H. Yu, G.-F. Hou, X.-W. Li, C.-Y. Ren and D.-S. Ma, *Polyhedron*, 2016, **112**, 61–66.

48 Y.-J. Ping, R.-Y. Yan, G.-C. Zhao, L. Qin and H.-G. Zheng, *Inorg. Chem. Commun.*, 2015, **61**, 136–139.

49 J. N. Babu Reddy, S. Vanishri, G. Kamath, S. Elizabeth, H. L. Bhat, D. Isakov, M. Belsley, E. de Matos Gomes and T. L. Aroso, *J. Cryst. Growth*, 2009, **311**, 4044–4049.

50 Z. Xue, H. Zhang, T. Sheng, Y. Wen, Y. Wang, S. Hu, H. Li, C. Zhuo and X. Wu, *Inorg. Chem. Commun.*, 2015, **55**, 99–102.

51 M.-F. Wu, G. Xu, F.-K. Zheng, Z.-F. Liu, S.-H. Wang, G.-C. Guo and J.-S. Huang, *Inorg. Chem. Commun.*, 2011, **14**, 333–336.

52 G.-X. Liu, X.-F. Wang and H. Zhou, *J. Solid State Chem.*, 2013, **199**, 305–316.

53 L. Guan and Y. Wang, *J. Solid State Chem.*, 2015, **230**, 243–248.

54 C. Yang, J. Wang, W. Wang and W. Zhan, *J. Solid State Chem.*, 2011, **184**, 2485–2489.

55 Y.-T. Wang, G.-M. Tang, Y. Wu, X.-Y. Qin and D.-W. Qin, *J. Mol. Struct.*, 2007, **831**, 61–68.

56 X.-F. Huang, Y.-H. Li, Q. Wu, Q. Ye and R.-G. Xiong, *Inorg. Chim. Acta*, 2005, **358**, 2097–2100.

57 H. Lun, X. Li, X. Wang, H. Li, Y. Li and Y. Bai, *J. Mol. Struct.*, 2017, **1127**, 662–667.

58 L.-Z. Chen, D.-D. Huang, J.-Z. Ge and F.-M. Wang, *Inorg. Chim. Acta*, 2013, **406**, 95–99.

59 Z.-X. Xu, Y. Liu and J. Zhang, *Inorg. Chem. Commun.*, 2016, **67**, 44–46.



60 M. Li, J. Yang, Y.-Y. Liu and J.-F. Ma, *Dyes Pigm.*, 2015, **120**, 136–146.

61 Z.-L. Zhang, X.-Q. Yao, N. An, H.-C. Ma, Z.-Q. Lei and J.-C. Liu, *Inorg. Chem. Commun.*, 2014, **45**, 127–130.

62 J. Zhu, H. Song, J. Sun, P. Yan, G. Hou and G. Li, *Synth. Met.*, 2014, **192**, 29–36.

63 L. Li, Z. Wang, Q. Chen, X. Zhou, T. Yang, Q. Zhao and W. Huang, *J. Solid State Chem.*, 2015, **231**, 47–52.

64 Y.-H. Luo, L.-L. Gu, X.-Y. Yu, F.-X. Yue, X. Chen and H. Zhang, *Inorg. Chem. Commun.*, 2014, **40**, 176–180.

65 P.-F. Wang, M.-G. Sheng, X.-S. Wu and X. Wang, *Inorg. Chim. Acta*, 2011, **379**, 135–139.

66 X.-F. Wang and G.-X. Liu, *Inorg. Chim. Acta*, 2013, **406**, 223–229.

67 L. Li, D. Sun, Z. Wang, X. Song and S. Sun, *Solid State Sci.*, 2009, **11**, 1040–1043.

68 Q.-Y. Chen, Y. Li, F.-K. Zheng, W.-Q. Zou, M. F. Wu, G. C. Guo, A. Q. Wu and J.-Sh. Huang, *Inorg. Chem. Commun.*, 2008, **11**, 969–971.

69 S. D. Huang, R.-G. Xiong, J. Han and B. R. Weiner, *Inorg. Chim. Acta*, 1999, **294**, 95–98.

70 Y. Zhao, L. Luo, C. Liu, M. Chen and W.-Y. Sun, *Inorg. Chem. Commun.*, 2011, **14**, 1145–1148.

71 Y.-H. Tan, J.-X. Gao, Z.-F. Gu, Q. Xu, C.-S. Yang, Y.-S. Luo, B. Wang, Y.-Z. Tang and H.-R. Wen, *Polyhedron*, 2015, **101**, 239–243.

72 D. Marabello, P. Antoniotti, P. Benzi, C. Canepa, E. Diana, L. Operti, L. Mortati and M. P. Sassi, *J. Mater. Sci.*, 2015, **50**, 4330–4341.

73 L.-N. Duan, Q.-Q. Dang, C.-Y. Han and X. M. Zhang, *Dalton Trans.*, 2015, **44**, 1800–1804.

74 D.-K. Bucar, G. S. Papaefstathiou, T. D. Hamilton, Q. L. Chu, I. G. Georgiev and L. R. MacGillivray, *Eur. J. Inorg. Chem.*, 2007, 4559–4568.

75 E. R. Parnham and R. E. Morris, *Acc. Chem. Res.*, 2007, **40**, 1005–1013.

76 G. Nickerl, A. Henschel, R. Grunker, K. Gedrich and S. Kaskel, *Chem. Ing. Tech.*, 2011, **83**, 90–103.

77 S. Mendiratta, C.-H. Lee, S.-Y. Lee, Y.-C. Kao, B.-C. Chang, Y.-H. Lo and K.-L. Lu, *Molecules*, 2015, **20**, 8941–8951.

78 L.-Z. Chen, Y. Huang, R.-G. Xiong and H.-W. Hu, *J. Mol. Struct.*, 2010, **963**, 16–21.

79 B. Chen and G. Qian, *Metal-Organic Frameworks for Photonics Applications*, Springer, Berlin, Heidelberg, 2014.

80 C. Wang, T. Zhang and W. Lin, *Chem. Rev.*, 2012, **112**, 1084–1104.

81 C. Pettinari, A. Tăbăcaru and S. Galli, *Coord. Chem. Rev.*, 2016, **307**, 1–31.

82 R. Ahlers and U. Ruschewitz, *Solid State Sci.*, 2009, **11**, 1058–1064.

83 J. D. Lin, X. F. Long, P. Lin and S. W. Du, *Cryst. Growth Des.*, 2010, **10**, 146–157.

84 J. D. Lin, S. T. Wu, Z. H. Li and S. W. Du, *Dalton Trans.*, 2010, **39**, 10719–10728.

85 D. Shaowu and Z. Huabin, *Struct. Bonding*, 2014, 145–166.

86 Q. Ye, Y.-H. Li, Y.-M. Song, X.-F. Huang, R.-G. Xiong and Z. Xue, *Inorg. Chem.*, 2005, **44**, 3618–3625.

87 P. A. Savarimuthu, *Inorg. Chem. Commun.*, 2008, **11**, 791–794.

88 F. Sun and G. Zhu, *Inorg. Chem. Commun.*, 2013, **38**, 115–118.

89 F. Wang, Y.-X. Tan, H. Yang, Y. Kang and J. Zhang, *Chem. Commun.*, 2012, **48**, 4842–4844.

90 H.-N. Wang, X. Meng, C. Qin, X.-L. Wang, G.-S. Yang and Z.-M. Su, *Dalton Trans.*, 2012, **41**, 1047–1053.

91 O. R. Evans, R. G. Xiong, Z. Wang, G. K. Wong and W. Lin, *Angew. Chem., Int. Ed.*, 1999, **38**, 536–538.

92 D. W. Fu, W. Zhang and R. G. Xiong, *Dalton Trans.*, 2008, 3946–3948.

93 W. Lin, L. Ma and O. R. Evans, *Chem. Commun.*, 2000, 2263–2264.

94 W. Lin, Z. Wang and L. Ma, *J. Am. Chem. Soc.*, 1999, **121**, 11249–11250.

95 O. R. Evans and W. Lin, *Chem. Mater.*, 2001, **13**, 2705–2712.

96 H. Zhao, Z.-R. Qu, H.-Y. Ye and R.-G. Xiong, *Chem. Soc. Rev.*, 2008, **37**, 84–100.

97 M. F. Wu, G. Xu, F. K. Zheng, Z. F. Liu, S. H. Wang, G. C. Guo and J. S. Huang, *Inorg. Chem. Commun.*, 2011, **14**, 333–336.

98 L.-Z. Wang, Z.-R. Qu, H. Zhao, X.-S. Wang, R.-G. Xiong and Z.-L. Xue, *Inorg. Chem.*, 2003, **42**, 3969–3971.

99 R. G. Pearson, *J. Am. Chem. Soc.*, 1969, **91**, 4947–4955.

100 R. G. Pearson, *J. Mol. Struct.: THEOCHEM*, 1983, **103**, 25–34.

101 P. A. Maggard, C. L. Stern and K. R. Poeppelmeier, *J. Am. Chem. Soc.*, 2001, **123**, 7742–7743.

102 X. A. Chen, X. Chang, H. Zang, Q. Wang and W. Q. Xiao, *J. Alloys Compd.*, 2005, **396**, 255–259.

103 E. O. Chi, K. M. Ok, Y. Porter and P. S. Halasyamani, *Chem. Mater.*, 2006, **18**, 2070–2074.

104 X. A. Chen, L. Zhang, X. Chang, H. Xue, H. Zang, W. Q. Xiao, X. Song and H. Yan, *J. Alloys Compd.*, 2007, **428**, 54–58.

105 H. Y. Chang, S.-H. Kim, P. S. Halasyamani and K. M. Ok, *J. Am. Chem. Soc.*, 2009, **131**, 2426–2427.

106 H.-Y. Chang, S.-H. Kim, K. M. Ok and P. S. Halasyamani, *J. Am. Chem. Soc.*, 2009, **131**, 6865–6873.

107 C. Sun, T. Hu, X. Xu and J. G. Mao, *Dalton Trans.*, 2010, **39**, 7960–7967.

108 B. Yang, C. Hu, X. Xu, C. Sun, H. Z. Jian and J. G. Mao, *Chem. Mater.*, 2010, **22**, 1545–1550.

109 D. W. Lee, S.-J. Oh, P. S. Halasyamani and K. M. Ok, *Inorg. Chem.*, 2011, **50**, 4473–4480.

110 E.-a. Kim, D. W. Lee and K. M. Ok, *J. Solid State Chem.*, 2012, **195**, 149–154.

111 M. Joffre, D. Yaron, R. J. Silbey and J. Zyss, *J. Chem. Phys.*, 1992, **97**, 5607–5615.

112 W. Lin, Z. Wang and L. Ma, *J. Am. Chem. Soc.*, 1999, **121**, 11249–11250.

113 Y. Liu, G. Li, X. Li and Y. Cui, *Angew. Chem., Int. Ed.*, 2007, **46**, 6301–6304.

114 O. R. Evans and W. Lin, *Chem. Mater.*, 2001, **13**, 2705–2712.

115 O. R. Evans, R. G. Xiong, Z. Y. Wang, G. K. Wong and W. Lin, *Angew. Chem., Int. Ed.*, 1999, **38**, 536–538.



116 W. Lin, L. Ma and O. R. Evans, *Chem. Commun.*, 2000, 2263–2264.

117 O. R. Evans and W. Lin, *Acc. Chem. Res.*, 2002, **35**, 511–522.

118 H. Zhang, X. Wang, K. Zhang and B. K. Teo, *Coord. Chem. Rev.*, 1999, **183**, 157–195.

119 Y. F. Zhou, D. Q. Yuan, B. L. Wu, R. H. Wang and M. C. Hong, *New J. Chem.*, 2004, **28**, 1590–1594.

120 P. Ayyappan, G. Sirokman, O. R. Evans, T. H. Warren and W. Lin, *Inorg. Chim. Acta*, 2004, **357**, 3999–4004.

121 B. Luther-Davies and M. Samoc, *Curr. Opin. Solid State Mater. Sci.*, 1997, **2**, 213–219.

122 J.-P. Zou, Q. Peng, Zh. Wen, G.-Sh. Zeng, Q.-J. Xing and G.-C. Guo, *Cryst. Growth Des.*, 2010, **10**, 2613–2619.

123 Z. Sun, T. Chen, N.-n. Cai, J.-w. Chen, L. Li, Y. Wang, J. Luo and M. Hong, *New J. Chem.*, 2011, **35**, 2804–2810.

124 A. V. Vinogradov, H. Zaake-Hertling, A. S. Drozdov, P. Lönnecke, G. A. Seisenbaeva, V. G. Kessler, V. V. Vinogradov and E. Hey-Hawkins, *Chem. Commun.*, 2015, **51**, 17764–17767.