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## REVIEW

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## Dual-atom catalysts with microenvironment regulation for water electrolysis

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Dual-atom catalysts (DACs) have emerged as highly promising and efficient catalysts for water electrolysis, primarily due to their distinct dual-atom site effects. Optimizing the catalytic performance of DACs involves precise regulation of their microenvironment, encompassing elemental compositions, electronic distributions, geometric configurations, and coordination environment. Despite growing research interests, a comprehensive understanding of how these microenvironmental regulations impact DAC performance remains elusive. This review seeks to elucidate the influence of microenvironmental regulation on the efficiency of DACs in water electrolysis. We begin by categorizing DACs into three types based on their dual-atom configurations: homonuclear, heteronuclear, and molecular, followed by specific analyses of the different connection modes between dual-atom sites. Subsequently, we provide a comprehensive overview of the relationships between microenvironment regulation and the resulting properties of DACs, highlighting the elemental, electronic, geometric, and coordination effects. Additionally, we review recent research and theoretical advancements in water electrolysis applications of DACs under various pH conditions, demonstrating how microenvironment regulation enhances their catalytic performances. We conclude by discussing the challenges and prospects of DACs, offering insights into potential future research directions.

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## 1. Introduction

With the rapid growth of energy required by society, the fast-growing fossil fuel consumption has led to environmental pollution issues.<sup>1,2</sup> Consequently, producing sustainable, eco-friendly, and renewable energy sources has emerged as one of our society's foremost challenges.<sup>3</sup> Hydrogen is currently the most promising clean energy support. This is attributed to its production of only water without carbon dioxide or other harmful gases, coupled with its high energy conversion efficiency during the combustion process.<sup>4</sup> Water electrolysis is one of the most efficient methods to produce hydrogen, involving the hydrogen evolution reaction (HER) at the cathode and the oxygen evolution reaction (OER) at the anode. Water electrolysis offers several advantages, including its environmentally friendly nature and cost-effectiveness, making it a promising solution for addressing energy and environmental challenges.<sup>5,6</sup> Specifically, developing highly active and stable catalysts is crucial to improving efficiency and reducing the cost of hydrogen production by water electrolysis. Initially, precious metal catalysts such as platinum (Pt) and iridium (Ir) garnered significant attention due to their



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exceptional catalytic activity and stability. However, their widespread application is impeded by their high cost and scarcity.<sup>7,8</sup>

One of the practical approaches to address this problem is to downsize the active metal catalysts from bulk to atomically dispersed atoms while maintaining their excellent electrocatalytic performance.<sup>9,10</sup> Atomically dispersed catalysts generally offer specific coordination microenvironments and optimal active sites, maximizing atomic efficiency and preserving crucial catalytic properties.<sup>5,11,12</sup> Single-atom catalysts (SACs) with a single active center have garnered extensive and intensive research attention due to their outstanding metal dispersion, high atomic utilization, and excellent catalytic performance.<sup>13–15</sup> However, SACs also have some drawbacks, such as low metal loading and isolated active sites, which significantly limit their diverse catalytic applications.<sup>16,17</sup> Moreover, as research on bifunctional reaction catalysis progresses, SACs with a fixed adsorption mode can potentially hinder their multiatomic catalytic pathways.<sup>18</sup> Recently, dual-atom catalysts (DACs) have been further developed and functionalized based on SACs.<sup>19,20</sup> DACs are catalysts composed of two metal atoms coordinated in a specific configuration on a support material.<sup>21–23</sup> The interactions between these two centers give each metal atom a distinctive electronic and atomic structure, setting them apart from their isolated single-atom counterparts. The two active sites synergistically regulate the activation and adsorption energies of reaction intermediates.<sup>24,25</sup> DACs possess several advantages and overcome drawbacks of SACs. For example, DACs can anchor more metal atoms at a single adsorption site than SACs, allowing for higher metal atom loadings. Besides, DACs provide adjacent unsaturated coordination sites capable of absorbing multiple reactants simultaneously, breaking the catalytic linear relationships on conventional SACs during reactions.<sup>16</sup> This feature facilitates various adsorption modes for key intermediates, optimizing reaction pathways and significantly boosting catalytic activity and selectivity. However, designing efficient DACs is particularly challenging due to different possible combinations of dual metal sites, along with the complexities of precisely controlling these sites and their interactions.

Within DACs, interactions manifest among adjacent metal atoms and between these metal centers and their substrates. These interactions are subject to various influences, such as elemental composition, electronic properties, geometric configurations, and coordination effects. Significantly, these factors' impact depends on the microenvironment surrounding the active sites.<sup>26,27</sup> This precise regulation of interactions within the interconnected active sites in DACs is known as microenvironmental regulation.<sup>16,26,28</sup> This regulation optimizes the overall catalytic performance, leading to significant improvements in the efficiency and durability of hydrogen production through electrochemical water electrolysis. However, while published reviews often focus on DACs with homo- or hetero-centers, and are mainly focused on their synthetic methods and catalytic applications, how two metal active centers synergistically facilitate the water electrolysis process has not been comprehensively summarized.

Herein, we provide an overview of the microenvironment regulation of DACs for electrochemical water electrolysis over

recent years. This review includes a classification of DACs based on different metal centers and discusses the microenvironment regulation by analyzing the site effects between dual-atom sites, as well as the corresponding structure–performance relationships through both the theoretical and experimental findings. Firstly, DACs will be discussed in homonuclear, heteronuclear, and molecular three categories, followed by specific analyses of the different connection modes between dual-atom units. Subsequently, we provide insights into how microenvironmental regulation influences their catalytic properties by investigating the dual-atom site effect of DACs, including elemental, electronic, geometrical, and coordination effects. This regulation includes self-regulation by the metal atoms (elemental and electronic effects) and regulation by surrounding ligands or supporting materials (geometrical and coordination effects). Furthermore, a comprehensive overview will be provided on the advanced catalytic applications of DAC in water electrolysis reactions, including HER, OER, and overall water splitting under different pH environments. A comprehensive analysis summarizes a range of DACs suitable for water electrolysis under specific conditions. Finally, the summary, challenges, and prospects for developing DACs are presented. An in-depth investigation into the development of DACs and their microenvironmental regulation will contribute significantly to advancing the comprehension of DACs. Additionally, it will provide guidance and references for further research and the application of DACs in catalysis.

## 2. Categories of DACs

Dual-metal sites in DACs exhibit various configurations, including direct metal bonding, heteroatom bridging (such as N, O, P, S, *etc.*), and two independently isolated metal sites. These arrangements can host dual metal atoms as homonuclear pairs, heteronuclear pairs, or molecular entities. As shown in Fig. 1, for homo- and heteronuclear, these specific structures are represented as “X3–M<sub>1</sub>–M<sub>1</sub>–X3”, “X3–M<sub>1</sub>–X–M<sub>1</sub>–X3”, “X3–M<sub>1</sub>–M<sub>2</sub>–X3”, “X3–M<sub>1</sub>–X–M<sub>2</sub>–X3” and “M<sub>1</sub>–X4 + M<sub>2</sub>–X4”, respectively, where M<sub>1</sub> and M<sub>2</sub> denote the central metal atoms, and X represents coordinated heteroatoms.<sup>16</sup> It should be highlighted that dual-atom molecular catalysts (DAMCs), which feature either homonuclear or heteronuclear sites, are discussed separately in this review due to their distinctive structure with functional groups. The synergistic interactions between homonuclear or heteronuclear dual-atoms are critically influenced by their microenvironments, thereby optimizing the catalytic kinetics and selectivity of the diatomic sites. This section categorizes DACs into three types: homonuclear, heteronuclear, and molecular, with an emphasis on their composition and structures. By providing concise introductions and examples, we summarized the distinctions and characteristics of these three categories of DACs.

### 2.1 Homonuclear DACs

Homonuclear DACs are a class of catalysts in which the active sites are composed of two identical metal atoms connected

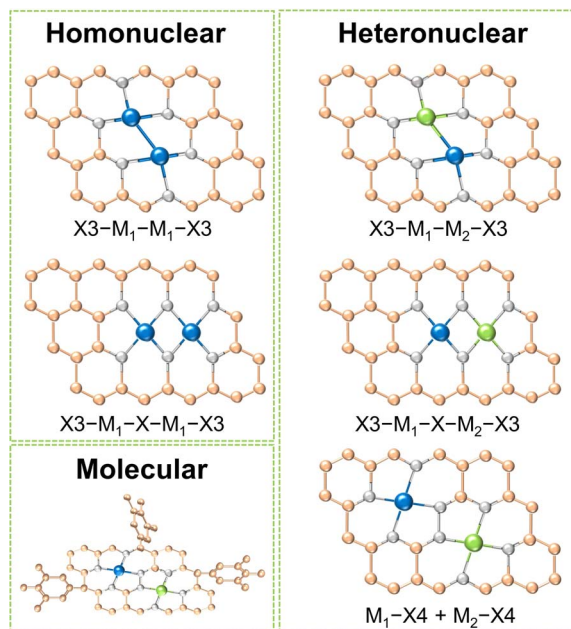


Fig. 1 Schematic structures of DACs.

through a metal bond or bridged by heteroatoms. By designing the metal centers and their coordination environments, researchers can tune the electronic structure and reactivity of homonuclear DACs to optimize their performance in specific reactions.<sup>9</sup> In this section, the discussion focuses on homonuclear DACs under the two connection modes, namely direct metal-bonded (“X3-M<sub>1</sub>-M<sub>1</sub>-X3”) and heteroatom bridged (“X3-M<sub>1</sub>-X-M<sub>1</sub>-X3”).

**2.1.1 Direct metal-bonded homonuclear DACs.** DACs with two sites connected directly by metal bonds are more easily synthesized. This is primarily due to the similar electronic structures and chemical properties of identical metal atoms, which enable the formation of strong metal-metal interactions. Direct metal bonding facilitates efficient electron transfer due to the strong interaction between two metal atoms, which has been demonstrated by many studies of homonuclear metal-bonded DACs in recent years.<sup>29–31</sup> For example, Zhang *et al.* used a cation vacancy-rich NiFeCo layered double hydroxide (NiFeCo-E) substrate to prepare a Pt@NiFeCo-E DAC with two types of Pt dual-atom active sites, *i.e.*, Pt dimer and Pt<sub>2</sub> pair (Fig. 2a).<sup>32</sup> The two types of Pt species have a percentage of 73.3% and 26.7% respectively, uniformly anchored on the NiFeCo-E substrate (Fig. 2b and c). Characterized by the Raman spectra (Fig. 2d), the authors demonstrate that the higher oxygen content and cationic vacancies of NiFeCo-E substrate are crucial for anchoring and forming Pt dual-atom sites. The formation of these vacancy-induced vibrations and the homogeneous Pt dual-atoms in Pt@NiFeCo-E suggest that the cationic vacancies serve as key anchoring sites for the Pt dual-atoms, preventing their aggregation into clusters. Similarly, Hao *et al.* synthesized Ni<sub>2</sub> DACs through high-temperature pyrolysis.<sup>33</sup> The Ni dual-atom sites were formed by the capture of evaporated Ni atoms in the carbon vacancies during

pyrolysis. The high-angle annular dark field scanning transmission electron microscopy (HAADF-STEM) (Fig. 2e and f) and extended X-ray absorption fine structure (EXAFS) (Fig. 2g) confirmed the presence of Ni-Ni dual-atom sites, with the Ni-Ni distance matching simulations (0.25 nm). The X-ray absorption near edge structure (XANES) spectrum (Fig. 2h) further elucidated the Ni coordination environment as Ni<sub>2</sub>N<sub>6</sub>. The studies demonstrate the successful synthesis of Pt and Ni DACs, in which the cationic vacancies of the substrate were crucial for anchoring and stabilizing these dual-atom active sites.

Except for anchoring and stabilizing the diatomic active site by creating substrate vacancies, utilizing specialized synthetic strategies to stabilize metal dimers is a viable approach based on the inherent tendency of metal atoms and clusters to polymerize. For example, Wei *et al.* adopt a “Heteroatom Modulator Approach” (HMA) to prevent the aggregation of iron atoms into nanoclusters by substituting Fe<sup>2+</sup> with other metal ions like Zn<sup>2+</sup> or Co<sup>2+</sup> (Fig. 2i).<sup>34</sup> The trinuclear complexes (Fe<sub>2</sub>Zn/Fe<sub>2</sub>Co) were immobilized on metal-organic frameworks (MOFs), followed by thermal evaporation of Zn atoms, resulting in a stable Fe<sub>2</sub>-GNCL (graphitized nitrogen-doped carbon layer). In Fig. 2j, the HAADF-STEM image and the accompanying intensity distribution map reveal a well-dispersed dual-site with a distance of approximately 2.6 Å on Fe<sub>2</sub>-GNCL. Besides, the Fe<sub>2</sub>-N<sub>6</sub> fraction with the lowest formation energies (*E<sub>f</sub>*) displays more excellent stability compared to the hypothesized Fe trimer cluster, indicating its optimal coordination configuration (Fig. 2k). In another study, Zhang *et al.* successfully synthesized supported Pd<sub>2</sub> DAC using the anion replacement deposition-precipitation (ARDP) method.<sup>35</sup> The atomic dispersion and metal-metal coordination structure of Pd<sub>2</sub> DAC were demonstrated using the aberration-corrected (AC) HAADF-STEM, XAFS, and X-ray photoelectron spectroscopy (XPS) imaging techniques.

**2.1.2 Heteroatomic bridged homonuclear DACs.** Compared to active sites with directly metal-bonded, introducing heteroatoms (such as O, S, P, N, *etc.*) to bridge dual-metal sites can alter the coordination structure. This adjustment leads to a redistribution of electrons at the metal active sites, thereby significantly enhancing the diversity of DAC coordination structures. It is worth noting that introducing heteroatoms is an effective strategy to prevent metal atom aggregation at the active sites.<sup>36</sup> Additionally, ensuring the interaction stability between active sites and heteroatoms is crucial. Zou *et al.* compared the stability and coordination interactions of homonuclear and heteronuclear directly bonded metal dimers (M<sub>1</sub>M<sub>2</sub>NC-3) with heteroatom-bridged metal dimers (M<sub>1</sub>M<sub>2</sub>NC-4), where M<sub>1</sub> and M<sub>2</sub> are two identical or different metal atoms (two configurations are displayed in Fig. 3a).<sup>37</sup> As shown in Fig. 3b, the computational of *E<sub>f</sub>* and average cohesive energy (Δ*E<sub>b-c</sub>*), elucidated enhanced stability of M<sub>1</sub>M<sub>2</sub>NC-4. The calculation results of Bader charges confirm that the M<sub>1</sub>M<sub>2</sub>NC-4 model bridged by heteroatoms has stronger coordination interactions. In another study, Chen *et al.* undertook comprehensive research on graphene-based N-bridge DACs (M<sub>1</sub>-N-M<sub>2</sub>/NC) by employing first-principle calculations and high-throughput screening strategies, the structure shown



Fig. 2 Structure characterization of homonuclear DACs connected by metal bonds. (a) Schematic diagram of the HER and OER activity of Pt dimers and Pt pairs on NiFeCo-E for water splitting catalysis. (b) Atomic-resolution HAADF-STEM image of Pt@NiFeCo-E. (c) A statistical diagram of Pt dimers and Pt<sub>2</sub>. (d) Raman spectra of NiFeCo, NiFeCo-E, Pt@NiFeCo, and Pt@NiFeCo-E<sup>32</sup> (reproduced with permission from ref. 32 copyright 2024, Elsevier). (e) HAADF-STEM image of Ni<sub>2</sub>NC, the dimers are depicted in red circles. Scale bar, 2 nm. Inset: the Ni–Ni atoms in the observed diatomic pair. Scale bar: 0.25 nm (area 1) and 0.5 nm (area 2). (f) Atomic distance of dual atoms in the experimental and simulated (Sim.) STEM images. Inset: The theoretical model is based on the simulated STEM image using quantitative STEM software. Scale bar, 0.25 nm. (g) FT of  $k^3$ -weighted Ni K edge EXAFS experimental data for Ni foil, NiO, NiPc, Ni<sub>1</sub>NC, and Ni<sub>2</sub>NC. (h) A comparison between the experimental XANES spectrum and the theoretical spectrum for Ni<sub>2</sub>NC. The main features reproduced are highlighted at point of A–E. The grey, turquoise, and purple spheres represent carbon, nitrogen, and nickel atoms, respectively. The Ni atoms with red circles mean the adsorbing atoms in the XANES spectrum<sup>33</sup> (reproduced with permission from ref. 33 copyright 2022, Springer Nature). (i) Heteroatom modulator approach for fabricating dual-atom iron catalysts on carbon layer (Fe: orange; Zn: purple). (j) HAADF-STEM image of Fe<sub>2</sub>-GNCL and intensity profiles (top and bottom) from the atomic sites 1 and 2. (k)  $E_f$  of Fe/Co–N<sub>4</sub>, Fe<sub>2</sub>–N<sub>6</sub>, and Fe<sub>3</sub>–N<sub>4</sub> moieties from graphene<sup>34</sup> (reproduced with permission from ref. 34 copyright 2020, John Wiley and Sons).

in Fig. 3c and d.<sup>38</sup> The  $E_f$  and binding energy ( $E_b$ ) calculation results of 14 homonuclear DACs consisting of transition metals confirm that most structures are thermodynamically stable (Fig. 3e). Therefore, the above theoretical calculation results confirm the enormous potential of heteroatom-bridged DAC.

Based on these theoretical guidances, Xu and co-workers recently successfully developed a highly dispersed Cu-based DAC using Cu<sub>2</sub>O as a precursor and hierarchical NC nanocage (hNCNC) as support.<sup>39</sup> As shown in Fig. 3f, HAADF-STEM imaging revealed the presence of Cu···Cu pairs with a spacing of 3.0–3.1 Å, indicating the formation of Cu···Cu–N<sub>4</sub> dual-metal sites and Cu–N<sub>4</sub> monometallic sites. These sites were facilitated due to the mesopores of *ca.* 0.6 nm and abundant nitrogen doping provided by h-NCNC. The random selection of four dual-metal sites revealed an average atomic distance of 3 Å, shown in Fig. 3g, longer than that in Cu crystals (2.556 Å), which confirms

the resemblance between twinned copper atomic sites and the Cu–O–Cu moieties present in Cu<sub>2</sub>O. The FT  $k^3$ -weighted EXAFS oscillation and the fitting of the EXAFS spectrum demonstrate the existence of Cu–O–Cu moieties in Cu<sub>1</sub>/hNCNC (Fig. 3h–i). Similarly, Wang *et al.* developed a general nanoparticle-to-single-atom-to-dual-atom (NP-to-SA-to-DA) aerosolization and sintering strategy for synthesizing the Co<sub>2</sub>N<sub>5</sub> dual atoms supported on nitrogen-doped HCS (Co<sub>2</sub>–N–HCS-900).<sup>40</sup> This approach was further proven to be effective in fabricating 22 different DACs. As a typical example, the AC HAADF-STEM image of Co<sub>2</sub>–N–HCS-900 (Fig. 3j) exhibited numerous uniformly distributed bright spots, representing Co atoms on the carbon framework. Notably, a significant portion of Co atoms were observed adjacent to each other (highlighted by red circles), forming Co<sub>2</sub> dual atoms with an approximate Co–Co distance of *ca.* 0.22 nm (Fig. 3k). As shown in Fig. 3l, the





**Fig. 3** Structure characterization of homonuclear DACs connected by heteroatomic bridges. (a) The top and side views of optimized  $M_1M_2NC-3$  and  $M_1M_2NC-4$ . (b)  $E_f$  and differences between the binding and cohesive energies of DACs ( $\Delta E_{b-c}$ )<sup>37</sup> (reproduced with permission from ref. 37 copyright 2022, RSC). (c) Structure of N-bridged  $M_1-N-M_2/NC$ . (d) 14 transition metals are taken into account in this work. (e)  $E_f$  and  $E_b$  of  $M_1-N-M_2/NC$ <sup>38</sup> (reproduced with permission from ref. 38 copyright 2023, Springer Nature). (f) HAADF-STEM image showing twinned Cu atoms (circled) and isolated Cu atoms. (g) Corresponding enlargement of the sites 1–4 in (f) and intensity profiles. (h) FT-EXAFS spectra of  $Cu_{-1}/hNCNC$  and the references. (i) FT-EXAFS fitting curve of the Cu K-edge in R-space for  $Cu_{-1}/hNCNC$  and the corresponding structure diagram<sup>39</sup> (reproduced with permission from ref. 39 copyright 2024, ACS). (j) AC HAADF-STEM images (Co dual atoms marked by red circles). (k) 3D atom-overlapping Gaussian-function fitting map in (j) for  $Co_2-N-HCS-900$ . (l)  $q$ -Space magnitude comparisons of Co foil,  $Co_{SA}-N-HCS-900$ , and  $Co_2-N-HCS-900$ . (m)  $k^3$ -Weighted EXAFS fitting curves at R space for  $Co_2-N-HCS-900$  (inset: schematic structure of  $Co_2N_5$  models; green: Co, blue: N, red: O)<sup>40</sup> (reproduced with permission from ref. 40 copyright 2023, Springer Nature).

comparison of single-atom and dual-atom sample curves in the FT-EXAFS fitting curve clearly shows the generation of Co–Co bonds during the formation of DA. In contrast, only the Co–N pathway was observed in  $Co_{SA}-N-HCS-900$ . The average coordination numbers of Co–N/O (3.9) and Co–Co (0.8) bonds in  $Co_2-N-HCS-900$  preliminarily suggested most Co atoms are bonded with 1 Co and 3 N (and 1 O) atoms. DFT calculations further confirmed the structure to be  $Co_2N_5-O$ , as illustrated in Fig. 3m.

## 2.2 Heteronuclear DACs

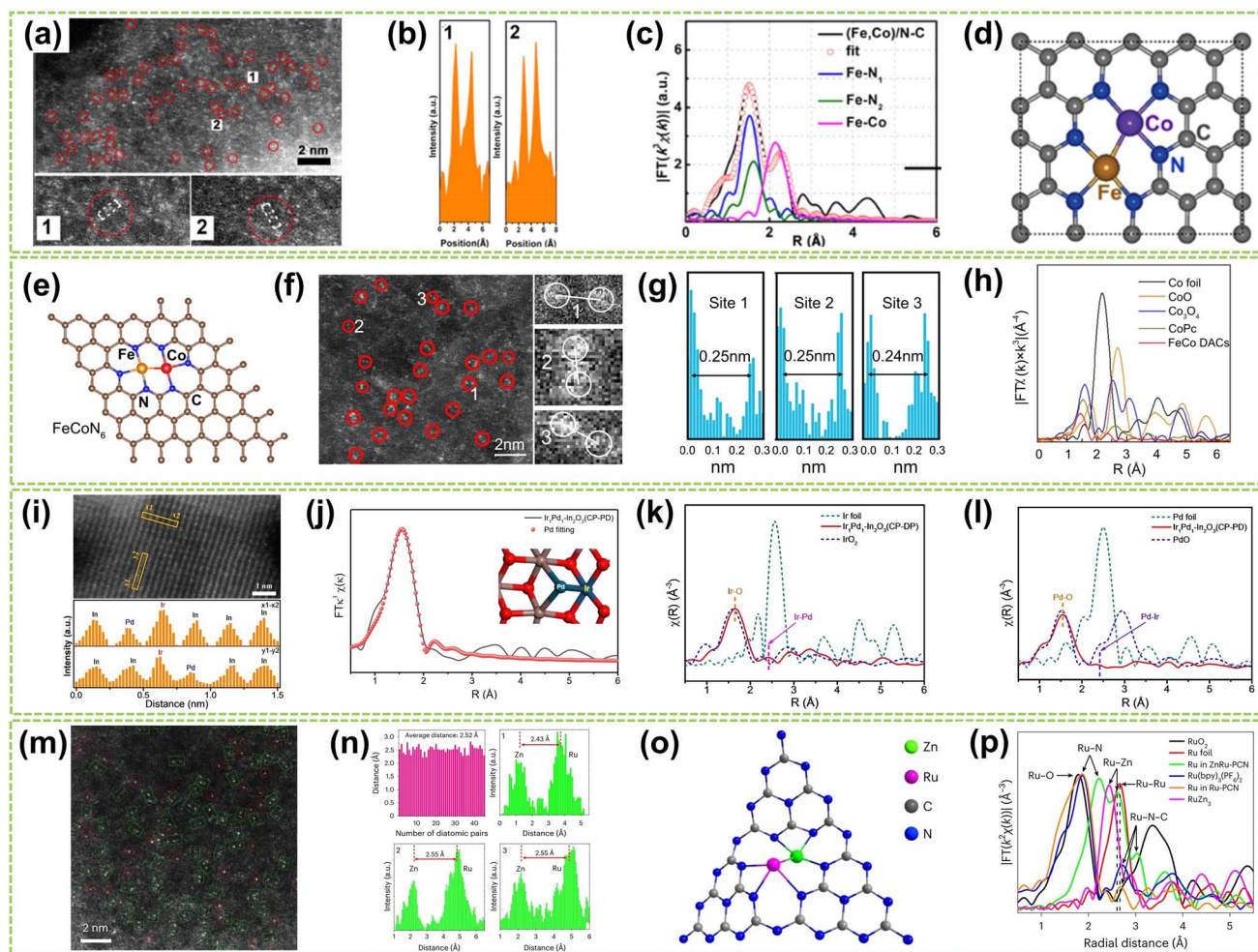
Heteronuclear DACs are a class of catalysts where the active sites consist of two distinct metal atoms.<sup>41</sup> Compared to

homonuclear DACs composed of two identical metal atoms, heteronuclear DACs feature different metal combinations, leading to a wide array of metal–metal interactions and electronic structures.<sup>42</sup> This diversity imparts unique catalytic properties to heteronuclear DACs, allowing for precise regulation of reactivity and selectivity in catalytic reactions.<sup>43</sup> Therefore, the catalytic performance of heteronuclear DACs is significantly enhanced. In this subchapter, heteronuclear DACs are classified into three categories: direct metal-bonded DACs (“ $X_3-M_1-M_2-X_3$ ”), heteroatom-bridged DACs (“ $X_3-M_1-X-M_2-X_3$ ”), and DACs with isolated sites (“ $M_1-X_4 + M_2-X_4$ ”).

**2.2.1 Direct metal-bonded heteronuclear DACs.** Similar to direct metal-bonded homonuclear DACs, heteronuclear DACs

connected by direct metal–metal bonds exhibit strong interactions that contribute to charge transfer in electrocatalytic reactions. However, different metals as active sites can offer more opportunities to enhance catalytic performance. Extensive research has focused on nonprecious metal-based DACs, especially Fe-based DACs, which have attracted significant attention because of their remarkable efficiency and cost-effectiveness. For instance, Wang and co-workers developed a host–guest strategy to synthesize Fe–Co dual sites embedded on N-doped porous carbon ((Fe, Co)/N–C).<sup>44</sup> They leverage a Zn/Co bimetallic metal–organic framework (BMOF) as a host to encapsulate FeCl<sub>3</sub> molecules within its cavities, and then using a two-solvent method, the Fe salts are fully immobilized inside the MOF's

porous structure. As shown in Fig. 4a and b, the HAADF-STEM images demonstrate the presence of Fe–Co dual sites, with the corresponding intensity profiles clearly distinguishing the dispersion of various Fe–Co dual sites. The FT  $k^3$ -weighted  $\chi(k)$ -function EXAFS results reveal two significant peaks at *ca.* 1.51 and 2.25 Å for (Fe, Co)/N–C, indicating that besides the main Fe–N path of (Fe, Co)/N–C DAC, the Fe–Co path also need to be considered in the surrounding coordination environment (Fig. 4c). Furthermore, using DFT calculation, the optimal structure of (Fe, Co)/N–C is shown in Fig. 4d, revealing the Fe–Co dual sites within the (Fe, Co)/N–C DAC through direct metal-bonded. In another work, Sun and co-workers successfully synthesized Fe–Co DACs by anchoring Fe–Co pairs on hollow



**Fig. 4** Structure characterization of heteronuclear DACs connected by metal bonds. (a) HAADF-STEM of (Fe, Co)/N–C, showing Fe–Co dual sites dominant in (Fe, Co)/N–C. (b) Corresponding intensity profiles were obtained on the zoomed areas in panel (a). (c) Fe K-edge EXAFS fittings of (Fe, Co)/N–C and reference samples. (d) Proposed architectures of Fe–Co dual sites<sup>44</sup> (reproduced with permission from ref. 44 copyright 2017, ACS). (e) The proposed model of Fe–Co DACs sites. (f) AC HAADF-STEM image of Fe–Co DACs and some Fe–Co diatomic sites are highlighted by red circles (the red circles 1, 2, and 3 are partially enlarged on the right). (g) Intensity profiles of the three sites in (f). (h) FT-EXAFS spectra of Co<sup>45</sup> (reproduced with permission from ref. 45 copyright 2023, Springer Nature). (i) The linear scanning profile along the region of the marked orange rectangles and the corresponding intensity of the atoms. (j) Pd fits of EXAFS spectra of the dual-atom Ir<sub>1</sub>Pd<sub>1</sub>–In<sub>2</sub>O<sub>3</sub> (CP-PD) catalyst (inset: the simulated dual-atom Pd–Ir configuration). EXAFS spectra of (k) Ir L-edge and (l) Pd K-edge over dual-atom Ir<sub>1</sub>Pd<sub>1</sub>–In<sub>2</sub>O<sub>3</sub> (CP-PD) catalyst<sup>46</sup> (reproduced with permission from ref. 46 copyright 2024, John Wiley and Sons). (m) AC HAADF-STEM image of ZnRu–PCN. (n) Statistics of diatomic distance and line-scanning intensity profiles obtained from (m), with the average distance being 2.5 Å. (o) The structures of ZnRu–PCN. (p) FT-EXAFS curves for ZnRu–PCN, Ru–PCN, and the references. *k*, *k* space;  $\chi(k)$ , the wavelength of *k* space<sup>47</sup> (reproduced with permission from ref. 47 copyright 2024, Springer Nature).



carbon spheres (HCS) using a two-step solvent impregnation method.<sup>45</sup> By adding an appropriate amount of Co ions onto the polydopamine (PDA) surface, the Co sites could then be leveraged as chemisorption centers to selectively adsorb polar iron species from the solution, enabling the accurate synthesis of Fe–Co bimetallic active sites. The schematic diagram of their structure is shown in Fig. 4e. The isolated Fe–Co pairs can be observed in the HAADF-STEM images, with an interatomic distance of approximately 0.25 nm (Fig. 4f and g). As shown in Fig. 4h, the structural spectroscopy obtained from XAFS also confirmed the Fe–Co–N<sub>6</sub> coordination environment by observing the peaks located at approximately 1.4 and 2.4 Å as the presence of Co–N and Co–Fe coordination in FeCo DACs. Moreover, Fan *et al.* recently combined a zeolitic imidazolate framework (ZIF) derivative strategy with subsequent thermal fixation treatment to prepare dual-metal Co<sub>1</sub>Fe<sub>1</sub>–N–C nanorods (Co<sub>1</sub>Fe<sub>1</sub>–NC–NRs) with atomically dispersed Co/Fe sites.<sup>48</sup> Similarly, Pan *et al.* successfully prepared Fe–Ni atomic dimers in N, P co-doped carbon-based catalysts through a simple *in situ* adsorption-thermal decomposition method.<sup>49</sup> The favorable configuration, with O\* on the Ni site and OH\* on the adjacent Fe site, combined with local electron engineering *via* P doping, significantly accelerates the rate-determining OOH\* formation step, which finally results in improved alkaline OER kinetics. Zhang *et al.* designed porous structures in graphene to generate numerous edge sites, which were further nitrified to anchor Fe/Cu atoms and form metal–metal dimers.<sup>50</sup>

Although transition metal-based DACs demonstrate outstanding catalytic properties, their stability is often compromised under harsh reaction conditions compared to the noble metal catalysts, underscoring the critical demand for advanced noble metal-based DACs. For example, Chen *et al.* employed photo-induced excitation to precisely position heteronuclear palladium atoms around the iridium atoms supported by indium oxide to form dual-metal Ir–Pd sites.<sup>46</sup> They employ light as the driving force. The short lifetime of the photo-generated electrons facilitates their rapid spillover and deposition near the single-atom site. This ensures that another atom can be precisely deposited near the location of the existing single atom, enabling the controlled fabrication of the desired diatomic configuration. In Fig. 4i, the different Z-contrasts between Ir/Pd atoms and the In<sub>2</sub>O<sub>3</sub> support were compared, and the corresponding atomic types were identified using line profiles. The EXAFS spectra of both Ir and Pd further confirm the above argument by the prominent peak at approximately 2.46 Å, which correlates with the length of the Ir–Pd bond in a model featuring In<sub>2</sub>O<sub>3</sub>-supported Ir–Pd dual-atom sites (Fig. 4j and k). Furthermore, a model of In<sub>2</sub>O<sub>3</sub>-supported Ir–Pd structure was established through EXAFS fitting, in which the distance between Ir and Pd (2.5 Å) closely matches with the results of EXAFS and line profiles in HAADF-STEM (Fig. 4l). This finding confirms the existence of Ir–Pd dual-metal pairs within such a structure. In another work, Zhao and co-workers recently provided a comprehensive overview of heteronuclear DACs.<sup>47</sup> Polymer carbon nitride (PCN) was employed as the substrate, and the “navigate and localize” strategy was exemplified using ZnRu–PCN DAC. This strategy precisely constructed

a heteronuclear ZnRu–PCN DAC using electronic accumulation on zinc sites to connect the ruthenium centers. In Fig. 4m and n, numerous dual-atom sites (green-boxed region) with an average distance of approximately 2.52 Å between adjacent sites were observed, inferring the dimeric structure (Fig. 4o). FT-EXAFS conducted a further analysis of the microstructure. As shown in Fig. 4p, peaks around 2.61 Å and 2.24 Å in the FT-EXAFS spectrum were observed in ZnRu PCN, attributed to the Ru–Zn and Ru–N pathways, respectively. At higher *R* values, no appreciable peaks for Ru–Ru were observed, confirming that most Ru atoms exist in single-atom form and coordinate with the Zinc atom in ZnRu DAC.

**2.2.2 Heteroatomic bridged heteronuclear DACs.** Optimizing the physicochemical properties of heterogeneous DACs by incorporating heteroatomic bridges such as N, O, S, and P mirrors the improvement seen in heteroatomic bridged homonuclear DACs. These bridges serve as intermediates in such structures, facilitating electronic redistribution within DACs. This redistribution enhances control over interactions between catalytic sites and their coordination environment, ultimately influencing the catalytic selectivity and activity of the DACs.<sup>41,51</sup> For example, Li *et al.* reported an N-doped graphene-supported Co–N–Ni DAC through the polymerization pyrolysis method, and the optimal structure is shown in Fig. 5a.<sup>52</sup> During the synthesis process, Co and Ni species are uniformly anchored to the substrate by grinding. The AC HAADF-STEM images exhibited excellent dispersion of Co and Ni atoms, along with the formation of dual-metal sites in Fig. 5b. Notably, as shown in Fig. 5c and d, the peaks of Co–N and Ni–N in Co–N–Ni DAC both confirm that N atoms bridged the Co and Ni dual-atom sites. Besides, the absence of Co and Ni clusters in the synthesized DAC, as confirmed by the absence of corresponding peaks in the R-space EXAFS spectra of Co and Ni samples, also provided this. Zhao and co-workers developed a surface modification strategy, in which the Co<sup>2+</sup> and Ni<sup>2+</sup> ions are chelated and deposited as dispersed single atoms on Try–Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> substrates to construct a dual-atomic CoNi supported on Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> MXene (CoNi–Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>).<sup>53</sup> As shown in Fig. 5e and f, the HAADF-STEM image of CoNi–Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> shows several dual-atomic pairs highlighted by red rectangles, and the average interatomic distance (*ca.* 0.37 nm) was revealed by the intensity profiles. The FT infrared (FTIR) spectra image reveals the bonding structure of metal atoms, confirming that both Co atoms and Ni atoms are coordinated with N atoms in CoNi–Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> (Fig. 5g). Besides, the specific coordination structure of this DAC was also confirmed using XPS spectroscopy. As shown in Fig. 5h, both Co and Ni are primarily bonded to O/N, confirming that Ni atoms are indeed chelated by N atoms in the tryptophan molecule. It is worth noting that the intensities of the Ni–O and Co–O peaks in the two samples are higher than those of the Ni–N/Co–N peaks, indicating that some Ni and Co atoms are directly bonded to the surface-terminated oxygen atoms.

For heteroatomic bridged heteronuclear, besides CoNi DACs as mentioned above, there has been extensively reported on other types of heteroatomic bridging diatomic materials.<sup>54,55</sup> For instance, Zhang *et al.* synthesized ten different DACs using



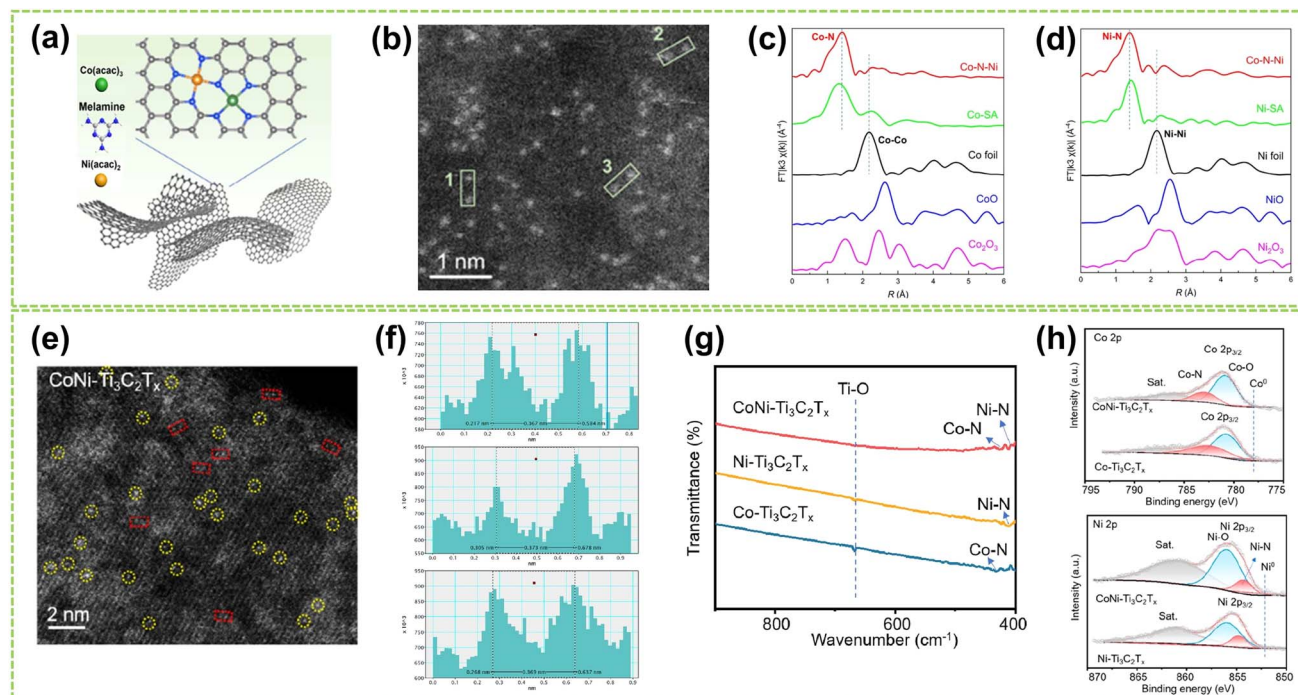
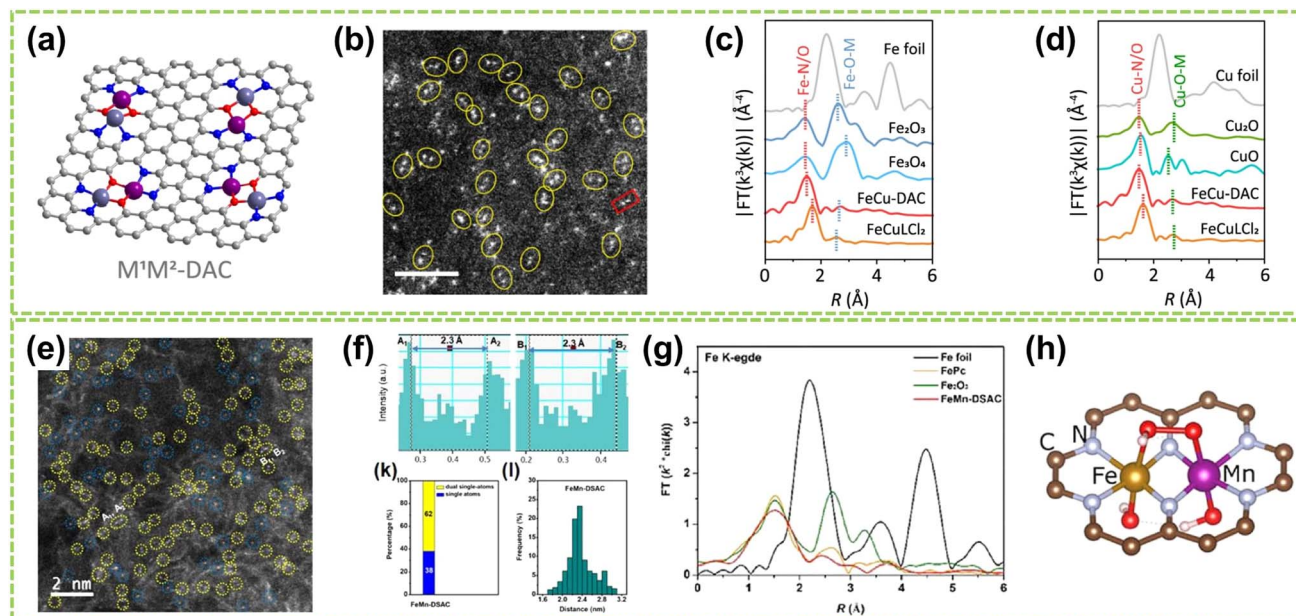


Fig. 5 Structure characterization of heteronuclear DACs connected by heteroatomic bridges. (a) Schematic diagram of the polymerization-pyrolysis synthesis method of Co-N-Ni. (b) AC HAADF-STEM image of Co-N-Ni. The bright spots in the green box represent the Co, Ni dual-atom. (c) Co K-edge and (d) Ni K-edge FT-EXAFS spectra of Co-N-Ni DAC, Co/Ni-SA, and the references<sup>52</sup> (reproduced with permission from ref. 52 copyright 2023, ACS). (e) HAADF-STEM images of CoNi-Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>, where bright spots corresponding to single metal atoms are marked by yellow circles, and possible Co-Ni atomic pairs are highlighted by red rectangles. (f) Intensity profiles were obtained on adjacent metal atomic pairs, taken from three different sites marked by red rectangles in (e). (g) FTIR spectra of CoNi-Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>, Ni-Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>, and Co-Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> composites. (h) Above: Co 2p and below: Ni 2p high-resolution XPS spectra of CoNi-Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> and Co-Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> composites<sup>53</sup> (reproduced with permission from ref. 53 copyright 2024, ACS).

a macrocyclic precursor-mediated approach, including six homonuclear species (Fe<sub>2</sub>, Co<sub>2</sub>, Ni<sub>2</sub>, Cu<sub>2</sub>, Mn<sub>2</sub>, and Pd<sub>2</sub>) and four heteronuclear species (Fe-Cu, Fe-Ni, Cu-Mn, and Cu-Co), as illustrated in Fig. 6a.<sup>56</sup> This approach allows the molecule to be embedded within ZIF-8, avoiding agglomeration during synthesis. As a typical example, the HAADF-STEM image of FeCu-DAC reveals a large amount of atomically dispersed dual-atom sites (Fig. 6b). Furthermore, using FT EXAFS spectra to confirm the coordination structure of DACs, the significant peak of FeCu-DAC in both Fe and Cu reveals the Fe-N/O path and Cu-N/O path are the main paths. Notably, the peaks observed at 2.57 Å in Fig. 6c can be attributed to Fe-O-M scattering, indicating the presence of Fe-O-M bonds in the system. Similarly, the peaks observed at 2.70 Å in Fig. 6d can be attributed to Cu-O-M scattering. These findings reveal that a heteroatomic bridge connects the FeCu dual-sites in FeCu-DAC. In another work, Cui *et al.* successfully prepared a Fe-Mn dual single-atom catalyst (DSAC) through a molten salt-assisted pyrolysis strategy.<sup>57</sup> HAADF-STEM images reveal the coexistence of single atoms (represented by blue dashed circles) and double single atoms (represented by yellow dashed circles) in FeMn-DSAC (Fig. 6e). A comprehensive statistical analysis of all identifiable bright spots, as shown in Fig. 6f, demonstrates that the proportion of dual single-atom moieties is 62%, with an atomic distance of approximately 2.3 Å. The local structural information around Fe/Mn detected by the FT of the *k*<sup>2</sup>-

weighted EXAFS spectra. Taking Fe K-edge spectra as an example, the absence of detectable metal-metal scattering peaks around 2.2 Å in FeMn-DSAC, as shown in Fig. 6g, indicates that the Fe and Mn sites are atomically dispersed in the two-dimensional nanosheets without a direct connection. Consequently, to ensure sufficient spatial proximity between Fe and Mn dual-atom sites and considering that the metal atoms are only connected to N atoms, the FeMn-DSAC model is shown in Fig. 6h. The measured distance between Fe and Mn sites in the model is approximately 2.3 Å, consistent with experimental results, confirming that Fe and Mn sites are bridged by heteroatom N. Furthermore, Yi *et al.* successfully synthesized low-cost CoCu bridged dual-atom center catalysts using a straightforward thermal decomposition method, allowing large-scale production in a single batch.<sup>58</sup> Shi and co-workers reported an N-bridged DAC composed of atomically dispersed indium and copper anchored on polymer carbon nitride (InCu/PCN),<sup>59</sup> which copper and indium are bonded to the PCN structure through metal-N coordination assemblies connected by Cu-N-In bridges. Similarly, a high-performance heteronuclear N-bridged Zn<sub>1</sub>Co<sub>1</sub> DAC<sup>60</sup> and a Cu-Fe dual-site electrocatalyst (CuFe DS/NC) with axial oxygen atom coordination<sup>61</sup> have been successfully synthesized.

**2.2.3 Heteronuclear DACs with isolated sites.** Apart from the configurations mentioned above, heteronuclear DACs with isolated atomic sites present another distinctive configuration.



**Fig. 6** Structure characterization of heteronuclear DACs connected by heteroatomic bridges. (a) Schematic of the localized structure of the  $M^1M^2$ -DAC. (b) HAADF-STEM images of FeCu-DAC. The scale bar in panels is 2 nm. FT EXAFS spectra of (c) Fe K-edge and (d) Cu K-edge for FeCu-DAC and other references<sup>56</sup> (reproduced with permission from ref. 56 copyright 2023, ACS). (e) AC HAADF-STEM images. The dotted circles in different colors show SAs (blue) or DSAs (yellow) in FeMn-DSAC. (f) Above: the corresponding Z-contrast analysis of regions A and B in (e). Below left: the statistical distribution of SAs and DSAs. Below right: statistical distribution of DSA distance. (g) FT of the  $k^2$ -weighted EXAFS spectra at Fe K-edge. (h) Structure diagram of FeMn-DSAC<sup>57</sup> (reproduced with permission from ref. 57 copyright 2022, John Wiley and Sons).

The two metal sites within the DAC are spatially separated without direct bonding, yet they still exhibit weak interactions facilitated by remote interactions or support materials. Notably, this spatial independence allows targeted catalysis for various reactions, mitigating competition between these two distinct sites. For example, Wang and co-workers successfully synthesized a dual-metal catalyst ( $Mn-Fe-N/S@mC$ ) featuring atomically dispersed  $Fe-N_4$  sites along with a small fraction of  $Mn-N_2S_2$  sites.<sup>62</sup> The precursor used for synthesizing  $Mn-Fe-N/S@mC$  is  $Mn_{0.43}Fe_{2.57}O_4$  nanocrystals, and the structure of  $Mn-Fe-N/S@mC$  DAC is illustrated in Fig. 7a. The FT  $k^3$ -weighted EXAFS curves further confirm the coordination environment of the metal atoms. As illustrated in Fig. 7b and c, two distinct peaks in the  $Mn-Fe-N/S@mC$  catalyst curves indicate the dominant Fe-N and Mn-N/S pathways. Furthermore, the absence of Fe-Fe/Mn-Mn peaks in the  $Mn-Fe-N/S@mC$  catalyst signifies the lack of metal clusters, confirming that the metal atoms are atomically dispersed. In another work, Wang *et al.* introduced boron to anchor  $FeN_4$  and  $NiN_4$  dual-atom sites on a porous carbon nanocage.<sup>63</sup> As shown in Fig. 7d, the introduction of boron-induced structural deformation led to forming “Fe-B-N” and “Ni-B-N” bridges. In addition, Chen *et al.* developed the “Ni- $N_4$ /GHSs/Fe- $N_4$  Janus” catalyst material, where single atoms Ni and Fe were incorporated into the inner and outer walls of graphene hollow nanospheres (GHSs), respectively.<sup>64</sup> The distinctive surface of the Janus material facilitates dual chemical reactions on a single surface, enabling the independent functionalization of different single atoms on distinct facets within the HGS (Fig. 7e and f). By utilizing the

least squares EXAFS fitting (Fig. 7g and h), the researcher investigated the structures of  $Ni-N_4$ /GHSs and GHSs/Fe- $N_4$ , revealing the coordination numbers of both Ni and Fe center atoms are approximately 4, with the bond lengths between these atoms and the surrounding coordinated N atoms are measured around 2 Å. This study demonstrated that the isolated active sites in DACs can effectively minimize mutual interference and achieve a balanced competition between different reactions. In another work, Wang *et al.* created a layered structure that tightly binds Ni and Fe atoms, forming axial dual-atom sites by aggregating defects.<sup>65</sup> The architecture provides three-dimensional confinement, preventing metal atoms from migrating from the plane and ensuring structural stability. AC HAADF-STEM imaging revealed vertically stacked multilayer graphene stripes with an interlayer spacing of 3.4 Å (Fig. 7i), and the majority of bright spots were located within the stacked graphene layers, forming axial dual-atom sites as illustrated in Fig. 7j. Moreover, FT EXAFS spectra of the Ni K-edge displayed a prominent peak at around 1.44 Å, indicating the Ni-N scattering path (Fig. 7k). The absence of Fe/Ni-Ni scattering peaks reveals that the appropriate interlayer spacing (3.4 Å) within the multilayer graphene effectively prevents metal atom bonding, resulting in atomic-level dispersion (Fig. 7l).

In addition to the Fe-based DACs (particularly Fe-Ni DACs) with isolated sites, Co-based and Cu-based DACs have also been successfully synthesized and extensively studied in recent years. For instance, Li *et al.* fabricated a DAC composed of copper and cobalt atoms on a highly porous NC substrate (Cu-Co/NC).<sup>66</sup> High-resolution AC HAADF-STEM imaging revealed paired



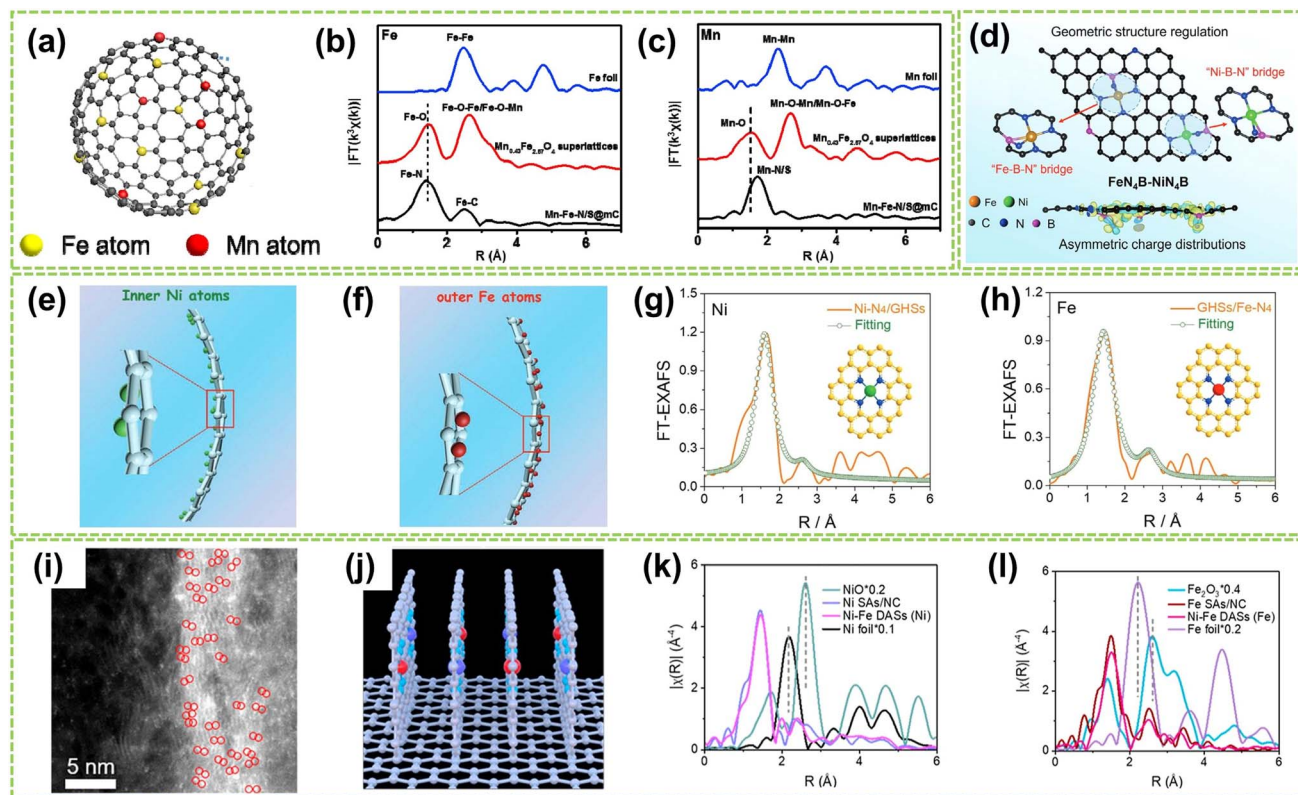


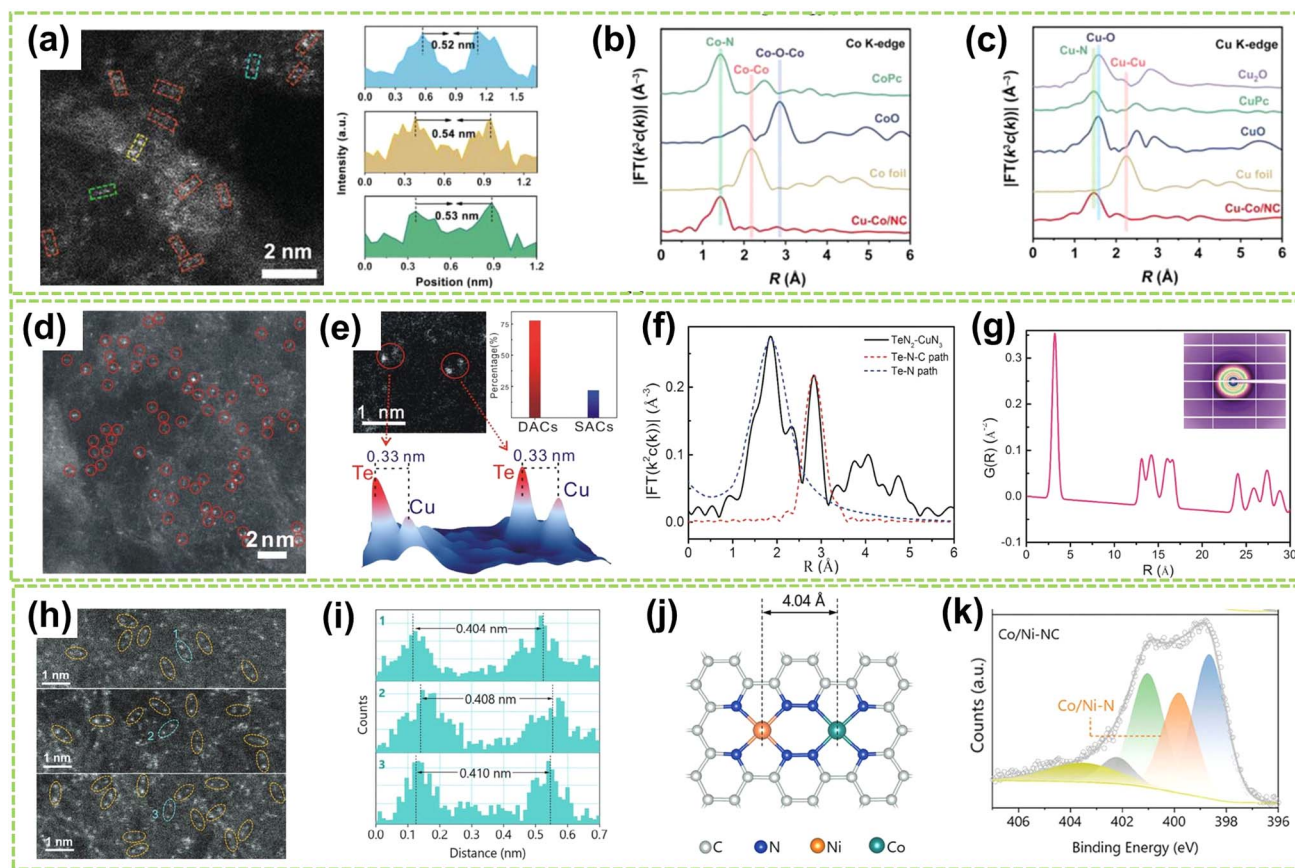
Fig. 7 Structure characterization of heteronuclear DACs with isolated sites. (a) Schematic illustration of Mn-Fe-N/S@mC catalysts. (b) Fe and (c) Mn  $k^3$ -weighted FT EXAFS curves of Mn-Fe-N/S@mC. The EXAFS curves of  $\text{Mn}_{0.43}\text{Fe}_{2.57}\text{O}_4$  supercrystals and metal foil are also included for comparison<sup>62</sup> (reproduced with permission from ref. 61 copyright 2019, Elsevier). (d) Illustration of atomic configurations after the introduction of boron in the first coordinated sphere of Fe-Ni dual atoms<sup>63</sup> (reproduced with permission from ref. 63 copyright 2024, John Wiley and Sons). Scheme of (e) the inner Ni atoms and (f) the outer Fe atoms. XAFS fitting curves of (g) Ni-N<sub>4</sub>/GHSs and (h) GHSs/Fe-N<sub>4</sub> R space (insets in (g) and (h): atomic structure model of Ni-N<sub>4</sub>/GHSs and GHSs/Fe-N<sub>4</sub>, respectively)<sup>64</sup> (reproduced with permission from ref. 64 copyright 2020, John Wiley and Sons). (i) AC HAADF-STEM image. The bright spots are highlighted by red circles. (j) Structural model of vertical graphene. FTs of  $k^3$ -weighted (k) Ni K-edge and (l) Fe K-edge FT-EXAFS spectra of Ni-Fe DASs, Ni SAs, and the references<sup>65</sup> (reproduced with permission from ref. 65 copyright 2023, ACS).

bright spots, with slightly brighter Cu spots and darker Co spots, indicating the presence of heterogeneous dual-atom sites in Cu-Co/NC DAC, with an average interatomic distance of 0.53 nm (Fig. 8a). Besides, FT-EXAFS spectra exhibited prominent peaks at 1.44 Å and 1.45 Å for CoPc and Cu-Co/NC, respectively, indicating the existence of Co-N coordination. The absence of metal-metal and metal-O-metal coordination peaks in Cu-Co/NC suggests that Co and Cu sites are atomically dispersed within the catalyst (Fig. 8b and c). In addition, Jiao and co-workers clarified the synergistic interaction between the dual active sites by synthesizing an asymmetric  $\text{TeN}_2\text{-CuN}_3$  DAC.<sup>67</sup> The formation of Te and Cu dual-atom sites was observed through HAADF-STEM image, where bright spots within red circles indicated the presence of dual-atom sites (Fig. 8d). Besides, clear identification of the dual-atom sites was achieved as shown in Fig. 8e, comprising approximately 78% of the total sites and the interatomic distance is approximately 0.33 nm. The significantly lower  $\chi(R)$  intensity observed from the Te-N-C pathway compared to the Te-Te pathway confirms the atomically dispersed distribution of Te on the support (Fig. 8f). Fig. 8g further confirms the presence of Cu-Te by

revealing a prominent peak at approximately 3.29 Å in the pair distribution function (PDF) Rietveld refinement results, corresponding to the Te-Cu pathway in  $\text{TeN}_2\text{-CuN}_3$ . Moreover, Yang *et al.* successfully synthesized Co/Ni-N-doped carbon catalysts by controlled pyrolysis of  $\text{Co}^{2+}$  and  $\text{Ni}^{2+}$ -containing compounds within ZIF-8.<sup>68</sup> As shown in Fig. 8h and i, the Co and Ni atoms are uniformly dispersed in  $\text{CoN}_4\text{-NiN}_4$  pairs with an approximate spacing of 0.41 nm. Therefore, the theoretical models of neighboring  $\text{CoN}_4$  and  $\text{NiN}_4$  sites simulated by density functional theory (DFT) calculations were obtained and illustrated in Fig. 8j, including the interatomic spacing (0.41 nm) and metal coordination environment with a coordination number of approximately 4. Fig. 8k reveals the pronounced presence of Co/Ni-N peaks within the Co/Ni-NC, further confirming the coordination environment of  $\text{CoN}_4\text{-NiN}_4$ .

### 2.3 Molecular DACs

Molecular catalysts possess the advantages of structural controllability and uniformity of active sites, making it easier to enhance the reactivity and selectivity of products through chemical modifications.<sup>26,69,70</sup> After Jasinski reported a cobalt



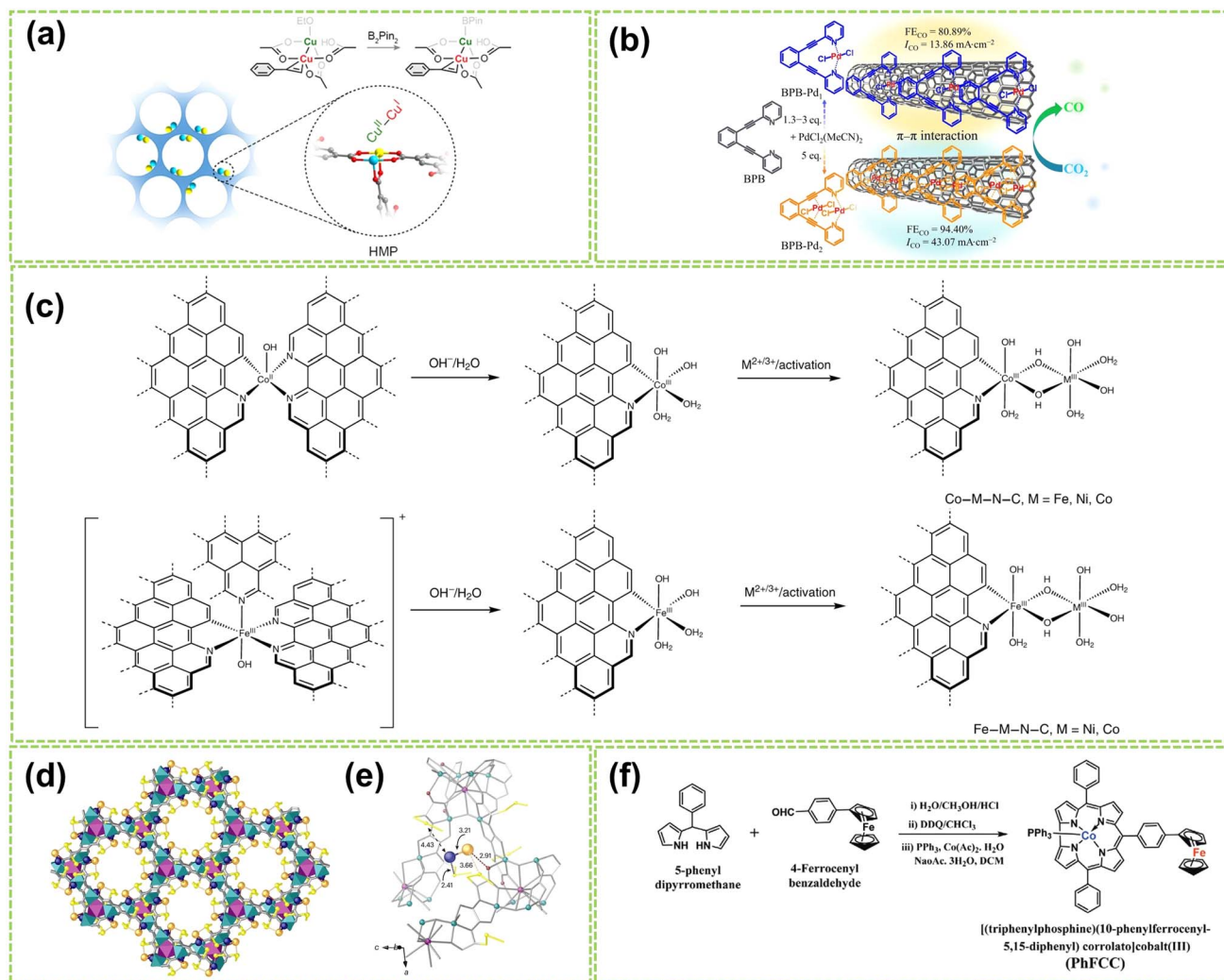
**Fig. 8** Structure characterization of heteronuclear DACs with isolated sites. (a) High-resolution AC HAADF-STEM image and the corresponding line-scanning intensity profiles. (b) Co K-edge and (c) Cu K-edge FT-EXAFS spectra of Cu–Co/NC and the references<sup>66</sup> (reproduced with permission from ref. 66 copyright 2023, John Wiley and Sons). (d) HAADF-STEM image of TeN<sub>2</sub>–CuN<sub>3</sub> DAC. (e) Above left: typical spot pairs for TeN<sub>2</sub>–CuN<sub>3</sub> DAC. Above right: statistical analysis for double sites and single sites. Below: corresponding 3D intensity profiles of two pairs of DAC sites. (f) Fitting results for Te–N and Te–N–C paths of TeN<sub>2</sub>–CuN<sub>3</sub> DAC. (g) PDF Rietveld refinement result for partial Cu–Te paths of TeN<sub>2</sub>–CuN<sub>3</sub> DAC. Inset: the 2D scattering image of DAC<sup>67</sup> (reproduced with permission from ref. 67 copyright 2023, Springer Nature). (h) AC HAADF-STEM images of Co/Ni–NC. (i) The corresponding diatomic distances are marked in (h). (j) Schematic model of neighboring CoN<sub>4</sub> and NiN<sub>4</sub> sites optimized by DFT calculations (the orange, green, blue, and gray spheres represent Ni, Co, N, and C atoms, respectively). (k) N 1s XPS spectra of Co/Ni–NC<sup>68</sup> (reproduced with permission from ref. 68 copyright 2024, John Wiley and Sons).

phthalocyanine-based electrocatalyst in 1964, the transition metal-based M–N<sub>4</sub> macrocyclic compounds have been found to have advantages such as the rich d-electron nature of transition metal centers and have now become one of the promising electrocatalysts.<sup>71</sup> Recent studies have revealed that Co and Fe-coordinated porphyrins, phthalocyanines, corrins, and corroles possess highly active catalytic sites and exhibit great potential as non-precious metal catalysts for the water electrolysis reaction.<sup>72–74</sup> DAMCs represent a class of molecular catalysts wherein two metal atoms serve as active centers. Compared to DACs, DAMCs provide the advantage of lower coordination numbers in carbon structure synthesis, significantly reducing the active site blockage.<sup>75</sup> This section focuses primarily on the structure of DAMCs to better understand the composition and arrangement of dual-atom active sites in molecular catalysts, thereby elucidating the interactions between reactants and catalysts.

For homonuclear DAMCs, Zhang and co-workers successfully prepared an ordered mesoporous catalyst with

heterovalent metal pairs (HMP) sites Cu<sup>II</sup>–Cu<sup>I</sup> on a microporous MOF (as schematically shown in Fig. 9a).<sup>76</sup> By controlling the distance between these sites (2.6 Å), avoiding the limitations imposed by micropores, thereby promoting the reaction. Besides, compared to the isovalent Cu<sup>II</sup>–Cu<sup>II</sup> sites, these Cu–HMP sites exhibited significantly improved expected yields. These findings provide important insights for future explorations of the performance of other HMP sites based on various MOFs. In another work, Zhang *et al.* synthesized two conjugated backbone dual-atom palladium complexes, BPB–Pd<sub>1</sub> SAC and BPB–Pd<sub>2</sub> DAC.<sup>77</sup> Due to the  $\pi$ -conjugated backbone of BPB, BPB–Pd<sub>1</sub> and BPB–Pd<sub>2</sub> catalysts are immobilized onto carbon nanotubes through  $\pi$ – $\pi$  stacking interactions, resulting in the formation of supported palladium complexes (Fig. 9b). Results and theoretical calculations indicate that the interaction between the dual-atom Pt species in BPB–Pd<sub>2</sub> DAC leads to a higher FE<sub>CO</sub> value. Besides, the electron effects generated between the sites reduce the free energy during the reaction process, thus facilitating the catalytic reaction. In another work,





**Fig. 9** Structure illustration of molecular DACs. (a) Schematic diagram and structural formula of the heterovalent metal pair (HMP) site of copper. The color scheme for the atom: cyan for Cu<sup>II</sup>, yellow for Cu<sup>I</sup>, red for O, and gray for C<sup>76</sup> (reproduced with permission from ref. 76 copyright 2024, ACS). (b) The preparation strategy for BPB-Pd<sub>1</sub> and BPB-Pd<sub>2</sub> and the electrocatalytic CO<sub>2</sub> reduction performance of catalysts supported on carbon nanotubes<sup>77</sup> (reproduced with permission from ref. 77 copyright 2024, Springer Nature). (c) A scheme showing the formation of DACs from the Co-based and Fe-based single-atom precatalysts. The dashed bonds connected to aromatic rings indicate extended N-doped carbon layers<sup>78</sup> (reproduced with permission from ref. 78 copyright 2021, Springer Nature). (d) Formation of Pd<sub>1</sub>-Au<sub>1</sub> dimers showing the structures of Pd<sub>1</sub>Au<sub>2</sub>@1 determined by SCXRD. Copper and strontium atoms from the network are represented by cyan and purple polyhedra. Organic ligands are depicted as grey sticks and thioether groups as yellow sticks in all cases. Orange and blue spheres represent gold and palladium atoms, respectively, in Pd<sub>1</sub>-Au<sub>1</sub> dimers. Blue, orange, and red dotted lines represent the Pd...S, Au...S, and Au...O interactions. (e) Details of X-ray crystal structure showing the main host-guest interactions and related structural parameters of Pd<sub>1</sub>-Au<sub>1</sub> dimers<sup>79</sup> (reproduced with permission from ref. 79 copyright 2024, Springer Nature). (f) Synthetic route for PhFCC<sup>80</sup> (reproduced with permission from ref. 80 copyright 2021, John Wiley and Sons).

the construction of a heteroatom-bridged Fe dimer site (Fe-O-O-Fe) is achieved by utilizing the face-to-face self-assembly of iron phthalocyanine (FePc) molecules under microwave irradiation for molecular O<sub>2</sub> adsorption.<sup>81</sup>

Furthermore, Bai *et al.* developed a general synthesis method for N-doped carbon-loaded DACs using *in situ* electrochemical conversion of single-atom precatalysts under OER conditions.<sup>78</sup> Six similar bimetallic catalysts containing Co, Fe, or Ni were synthesized by this method, and the synthesis process is displayed in Fig. 9f. This study reveals the formation of O-O bonds by nucleophilic attack on M(IV)=O with changes based on metal

ions and coordination. Among these DACs, the Co-Fe-N-C and Ni-Fe-N-C catalysts exhibited high TOF values for OER in alkaline conditions, which are close to or higher than 1 s<sup>-1</sup> at an overpotential of 300 mV. In addition, Ballesteros-Soberanas and co-workers prepared a uniform Pd<sub>1</sub>-Au<sub>1</sub> heteronuclear DAMC using a highly stable, crystalline MOF as a nanoreactor.<sup>79</sup> The precise atomic structure was revealed by single-crystal X-ray diffraction (SCXRD), as shown in Fig. 9c and d. The MOF's hexagonal channels (~0.6 nm) contain the Pd<sub>1</sub>-Au<sub>1</sub> dimer, stabilized by dimethyl sulfide chains. The Pd atoms are connected by 2.41 Å sulfur binding sites, while the Au atoms

interact with MOF oxygen atoms and are weakly sulfur binding. Moreover, Samireddi *et al.* successfully synthesized a heteronuclear dual-metal complex (PhFCC) of Co and Fe using multi-walled carbon nanotubes (MWCNTs).<sup>80</sup> As shown in Fig. 9e, the synthetic route of PhFCC involves the addition of phenyl groups, which extends the conjugated electron system and facilitates interactions with graphitic or electron-rich carbon supports.

In conclusion, DACs consist of homonuclear or heteronuclear metal atoms as active sites exhibit varied intrinsic structures and chemical environments. Homonuclear DACs typically exhibit excellent single-reaction catalytic activity due to the similar electronic structures and adsorption properties around the metal atoms. In contrast, the heterogeneous structure of heteronuclear DACs endows them with the capability to possess high HER and OER bifunctional catalytic performances simultaneously. Furthermore, by rational designing of metal components and coordination environments of DACs, the energy barrier of the key reaction steps in water electrolysis would be reduced due to the synergistic effect between two adjacent metal sites, thus improving the catalytic activity, selectivity, and stability.

### 3. Microenvironment regulation of dual-atom site effect in DACs

DACs perform superior to SACs due to multiple adsorption sites and the unique site effects between active sites. Therefore, microenvironment regulation of the effects between dual-atom sites significantly enhances the catalytic activity, stability, and selectivity of DACs.<sup>7,42</sup> This section comprehensively explains the mechanisms involved in the element, electronic, geometric, and coordination effects. It is worth noting that while the discussion has been categorized into four distinct aspects, these site effects generally work together in practical reactions, synergistically modulating the DAC microenvironment to enhance the catalyst's performance.

#### 3.1 Element effect

In DACs, the element effect can originate from three possible aspects: the interaction between the active metal centers, the incorporation of bridged heteroatoms, and the support atoms. Among them, the electronegativity of atoms affects the ability of central metal atoms to attract electrons, and higher electronegativity can lead to stronger adsorption of reactant molecules and intermediates, making it one of the critical factors in catalyst element selection.<sup>82,83</sup> Consequently, the electronegativity differences between sites promote electron transfer, optimizing the electronic structures and adsorption configurations of DACs, affects reaction kinetics and selectivity, thereby exhibiting excellent electrocatalytic activity.

For instance, Hao *et al.* conducted specialized research on interatomic electronegativity to demonstrate that shifts in electronegativity between Cu and Ni atoms result in significant electronic interactions in CuNi DAC.<sup>84</sup> The electronegativity compensation effect of adjacent metal atoms in the synthesized

CuN<sub>4</sub>-NiN<sub>4</sub> catalyst promotes the formation of intermediates during the reaction process, thereby improving the activity and selectivity of the catalyst (Fig. 10a). Furthermore, they found that the significant decrease in activation energy ( $E_a$ ) of CuN<sub>4</sub>-NiN<sub>4</sub> (0.08 eV) promotes CuNi dual-atom sites on reaction kinetics. In another work, Tong and co-workers synthesized Cu/Zn-NC dual-metal atom dispersed catalysts composed of Cu-N<sub>4</sub> and Zn-N<sub>4</sub> species on N-doped carbon supports.<sup>88</sup> Notably, the lower electronegativity of Zn atoms enables electron transfer to Cu atoms, facilitating O-O bond cleavage at the Cu-N<sub>4</sub> active sites, significantly enhancing the catalyst's overall reactivity. Besides selecting appropriate dual-atoms by comparing their electronegativity, DFT calculation prior to experimental works also provides an effective approach for identifying the DACs with versatile metal active sites conducive to high-efficiency catalysis.<sup>85,86,89,90</sup> For instance, Liu *et al.* employed DFT calculation to identify optimal dual-metal active sites for catalysis, comparing the predicted limit potential and potential-determining steps between Fe-based and Cu-based homo/heteronuclear DACs on C<sub>2</sub>N substrate.<sup>85</sup> As shown in Fig. 10b, the elemental compositions of DACs significantly impact the catalytic reaction pathway, thereby influencing the selectivity of the catalyst. Furthermore, Jiao and co-workers also performed preliminary screening of Co-M (M = Fe, Cu, Zn) DACs based on carbon paper (CP) substrates using DFT calculations.<sup>86</sup> As shown in Fig. 10c, a closer Gibbs free energy for H adsorption ( $\Delta G_{H^*}$ ) to zero under acidic and alkaline conditions indicates better hydrogen evolution performance. Experimental simulations of reaction mechanisms and comparison of relative free energies of intermediates confirmed the excellent HER activity of the prepared CoCu@CP DAC and CoFe@CP DAC under acidic and alkaline conditions, respectively (Fig. 10d and e). These results validate the ability of dual-atom centers to reduce the relative free energy effectively. Similarly, Liu *et al.* conducted DFT calculations to develop free energy profiles along the reaction pathway on Pt-Ru dimer structures and investigate the correlation between their activity of water-electrolysis reactions.<sup>90</sup>

Except for the interactions between metal atoms serving as active sites, the bridged heteroatoms doped in the dual-atom sites also induce similar elemental effects. Unlike the elemental metal-metal impact between different sites, the effects resulting from elements doping are typically metal-nonmetal elemental effects. Recently, researchers have modulated the electronic structure of dual-atom sites by doping with elements such as N, O, P, and S to regulate the microenvironment of DACs. Among them, doping N and O in graphite substrates to anchor metal sites is a common strategy to attain stabilization while ensuring the dispersion of atoms at the desired locations.<sup>16,91</sup> Moreover, previous research suggests that introducing elements with relatively lower electronegativity, such as S and P, offers a more precise approach to manipulating the electronic structure of dual-atom sites.<sup>92-95</sup> For example, Zhang *et al.* doped S on defect-rich porous nitrogen and sulfur carbon frameworks (DNSC) co-anchored with Fe and Ni.<sup>87</sup> The DFT-optimized structure is shown in Fig. 10f and g, showing that introducing S leads to forming carbon vacancies. This



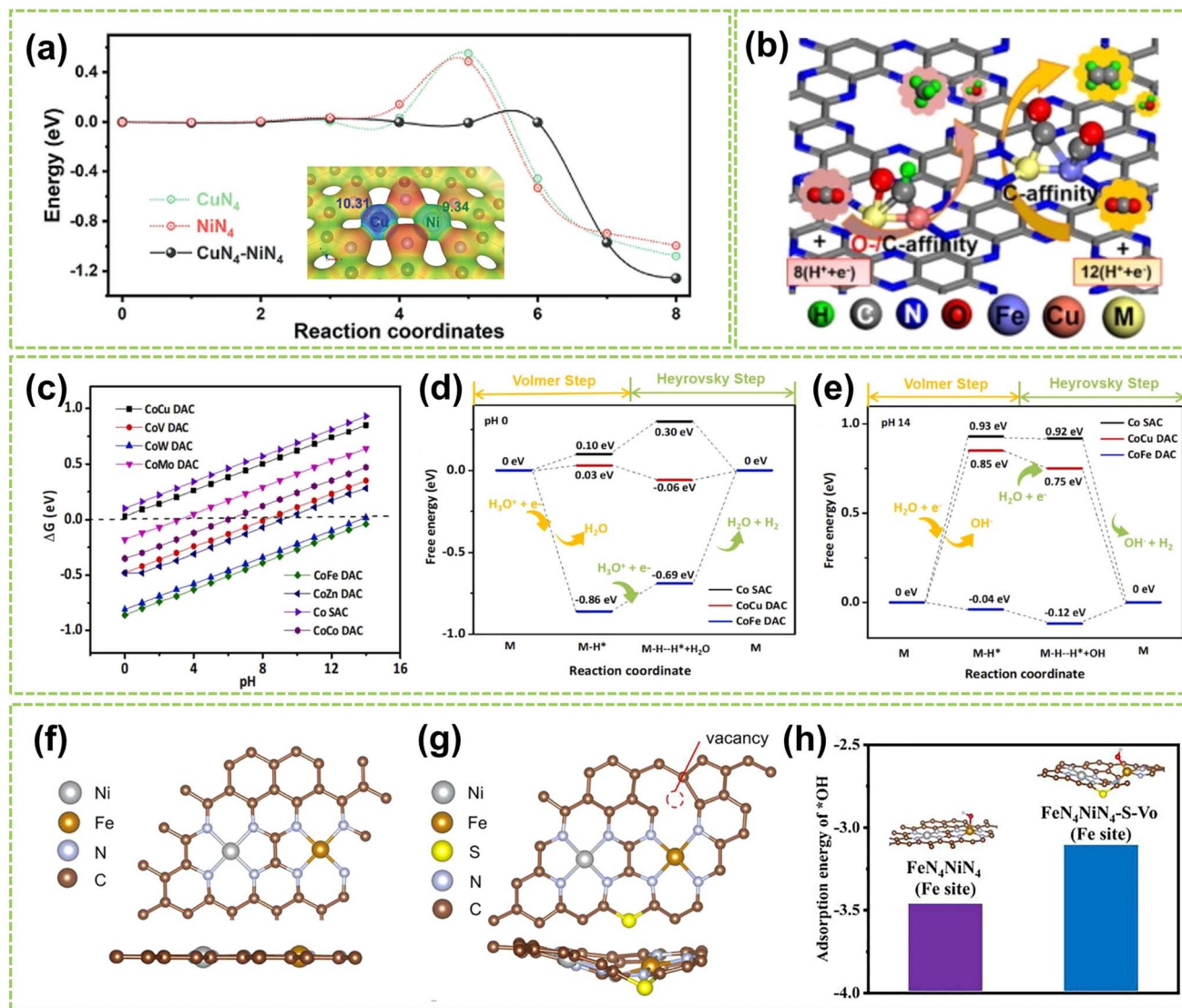


Fig. 10 Element effect of DACs. (a) CI-NEB-calculated kinetic reaction processes, containing the energy variation in the initial, transition, and final states toward  $^*\text{COOH}$  intermediate formation on  $\text{NiN}_4$ ,  $\text{CuN}_4$ , and  $\text{CuN}_4\text{-NiN}_4$ . Inset: electronegativity compensation in the Cu–Ni dual atom site<sup>84</sup> (reproduced with permission from ref. 84 copyright 2022, John Wiley and Sons). (b) Schematic illustration of DFT-optimized  $\text{M}_2\text{@C}_2\text{N}$  DACs performance<sup>85</sup> (reproduced with permission from ref. 85 copyright 2021, Elsevier). (c)  $\Delta G_{\text{H}^*}$  on Co SAC and different CoM DACs. Free energy diagram of CoCu DAC, CoFe DAC, and Co SAC under (d) acidic and (e) alkaline conditions, the “M” represents the CoCu DAC, CoFe DAC, and Co SAC, respectively<sup>86</sup> (reproduced with permission from ref. 86 copyright 2023, Elsevier). The optimized structure of (f)  $\text{FeN}_4\text{NiN}_4$  and (g)  $\text{FeN}_4\text{-NiN}_4\text{-S-Vo}$  (top view: above; side view: below). (h) The adsorption energy of  $^*\text{OH}$  on  $\text{FeN}_4\text{-NiN}_4$  and  $\text{FeN}_4\text{-NiN}_4\text{-S-Vo}$ <sup>87</sup> (reproduced with permission from ref. 87 copyright 2021, Elsevier).

phenomenon increases the d-electron density of the Fe site at the Fermi energy level and leads to electron redistribution. Furthermore, Fig. 10h illustrates that the binding strength between the  $\text{FeN}_4\text{-NiN}_4\text{-S-Vo}$  sites and intermediates is relatively weak, enhancing its catalytic performance. These findings reveal that introducing heteroatoms and carbon vacancies exerts a crucial influence on the performance of DACs.<sup>96</sup> Similarly, the study of Zhang and co-workers also validates that S doping in DACs enhances reaction kinetics.<sup>97</sup> In conclusion, regulating the element effects between various atom sites is a crucial method for controlling the microenvironment of DACs. This approach governs the central atoms' intrinsic

activity by coordinating various functional atoms. After that, the activity and stability of DACs are further enhanced through optimizing the adsorption of critical intermediates during the reaction process. Additionally, the electronic effects between sites and the coordination effect of support atoms are closely related to the elemental effects, which will be discussed in the following sections.

### 3.2 Electronic effect

In DACs, electronic effects are typically achieved through electron transfer between two adjacent metal atoms.<sup>17</sup> This phenomenon can be interpreted as one metal center acting as

the active site while the other is the electron regulator.<sup>16</sup> In addition to electron transfer, the electronic effect also involves electron sharing and redistribution, impacting the electronic structure, charge distribution, and stability of reaction intermediates. These factors are crucial in regulating the microenvironment of DACs and affecting the catalytic process. Recently, Wang *et al.* successfully controlled the transformation from Co nanoparticles to Co<sub>2</sub>N<sub>5</sub> dimers.<sup>40</sup> As shown in Fig. 11a, Co<sub>2</sub>N<sub>5</sub> dimers exhibit a significantly higher charge (1.11e) compare to other comparative catalysts. The experimental results confirm that the decrease in spin magnetic moment from nanoparticle to single-atom to dual-atom is primarily attributed to the redistribution of Co-3d orbital electrons due to level reordering, which weakens the adsorption of reaction intermediates and ultimately enhances the catalytic activity (Fig. 11b). In another work, Yang *et al.* synthesized Co/Ni-NC catalysts dispersed on an NC matrix.<sup>68</sup> Through the analysis of the charge density difference between Ni and Co, clear electronic delocalization between the metals and neighboring N atoms can be observed (Fig. 11c). This electronic delocalization facilitates the adsorption of O<sub>2</sub>, thereby enhancing the catalytic activity. In addition, Fig. 11d compares the free energy of intermediates during OER.

The relatively low adsorption energy of Co sites reveals that the Co sites in Co/Ni-NC serve as the active sites for the OER. In contrast, the NiN<sub>4</sub> sites improve the reactivity of adjacent CoN<sub>4</sub> active sites through electrostatic interactions. Consequently, electronic coupling between Co/Ni dimers can optimize the adsorption and desorption processes of atoms, molecules, and reaction intermediates, thereby enhancing the catalytic performance.

Moreover, Gao's group loaded atomic Pd on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>(012) to construct unitary Pd-Fe pairs.<sup>98</sup> As shown in Fig. 12a and b, an abundant delocalized charge accumulates along the Pd-Fe bonds, and the d-band of Fe closely matches that of Pd. These phenomena induce a strong electronic coupling in Pd-Fe pairs, which enhances the d-electron domination near the Fermi level and offers active sites for H<sub>2</sub> adsorption and activation. In addition, Chen *et al.* designed a DAC composed of Co-Fe double sites (Co/Fe-SNC).<sup>99</sup> Fig. 12c-e illustrate the investigation of bonding and antibonding orbitals near the Fermi energy level. In Co/Fe-SNC, the Co and Fe SA sites exhibit electron-rich, highly electroactive regions absent in the other two SNCs, revealing a strong orbital coupling. In addition, Fig. 12f-h show the Co-3d orbitals displaying a distinct peak close to the Fermi

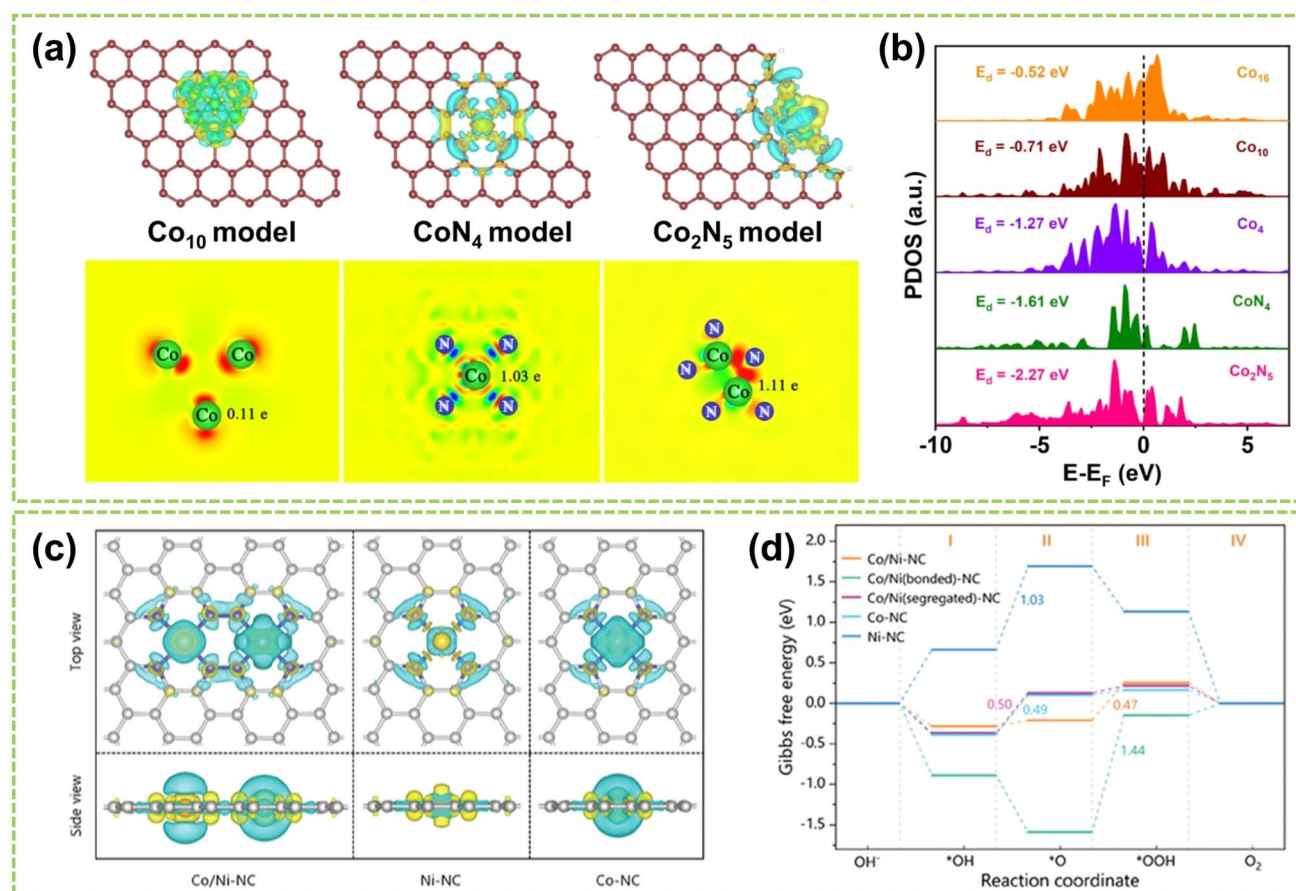


Fig. 11 Electronic effect of DACs. (a) Charge density difference and Bader charge diagrams. (b) Projected density of states (PDOS)<sup>40</sup> (reproduced with permission from ref. 40 copyright 2023, Springer Nature). (c) Charge density difference of Co/Ni-NC and the references (iso-surface = 0.008 a. u., yellow and blue regions represent electron accumulation and electron depletion, respectively). (d) Calculated  $\Delta G$  (in eV) profiles for OER processes on catalysts under alkaline conditions at  $U = 1.23$  V (ref. 68) (reproduced with permission from ref. 68 copyright 2024, John Wiley and Sons).





Fig. 12 Electronic effect of DACs. (a) Differential charge density (cyan and yellow represent charge depletion and accumulation, respectively; the cutoff of the density-difference isosurface is  $0.05 \text{ e } \text{\AA}^{-3}$ ). (b) PDOS for the total d orbitals of Pd and  $\text{Fe}^{98}$  (reproduced with permission from ref. 98 copyright 2022, ACS). 3D contour plot of bonding and anti-bonding orbitals near the Fermi level in (c) Co/Fe-SNC, (d) Co-SNC, and (e) Fe-SNC. Blue isosurface = bonding orbitals, and green isosurface = anti-bonding orbitals. PDOS of (f) Co/Fe-SNC, (g) Co-SNC, and (h) Fe-SNC. Pink balls = Co, orange balls = Fe, blue balls = N, yellow balls = S, and grey balls =  $\text{C}^{99}$  (reproduced with permission from ref. 99 copyright 2023, RSC).

level ( $E_v = 0 \text{ eV}$ ) at an  $E_v$  value of  $0.33 \text{ eV}$ , indicating their pronounced electrical activity in OER with a strong electron-depleting capability. The intense overlap between N-2p, C-2p, and S-3p orbitals also serves as electron-supplying centers, enabling efficient site-to-site electron transfer within Co/Fe-SNC. In another work, Wang *et al.* investigated N-doped graphene-supported (NGr) Ni-Fe DAC, with Ni acting as a nucleophilic center and Fe serving as an electrophilic center.<sup>100</sup> It is attributed to substantial charge transfer occurring upon the binding of Fe and Ni metal atoms to the NGr surface, resulting in the difference in positive charges. Therefore, the positively charged Fe exhibits a significantly higher potential to Ni, ultimately achieving the regulation of catalytic performance.

### 3.3 Geometric effect

Except for the two site effects mentioned above, the different geometric structures of sites in DACs also highly affect catalytic performance.<sup>101</sup> Generally, the geometric effect functions from two aspects: the distance between active metal centers and the spatial positioning of these centers. By changing the distance

between two active sites, the site distance effect is regulated, which will affect the adsorption of reactants, the process of electron transfer, and reaction kinetics, thereby impacting catalytic activity and selectivity. On the other hand, the spatial positioning effect of the active sites involves the arrangement of atoms and geometric configurations on the catalyst surface, which also significantly influence the reaction.

The distance effect is achieved by modifying the spacing between dual metal atoms or incorporating non-coordinated atoms to fine-tune their separation from the central atom.<sup>102–104</sup> For instance, Yao *et al.* investigated how the distance effect impacts the catalytic reaction performance of DACs by manipulating the spacing between the dual-metal atoms.<sup>103</sup> They used dispersed NiCu-NC as an example to construct DAC models with four different site distances. Subsequently, DFT calculations then quantified the site distances and net electron difference, illustrating the electron redistribution effect. As depicted in Fig. 13a, the net electron difference gradually decreases as the distance widens, becoming trivial when the separation surpasses  $5.0 \text{ \AA}$ , at which point the site distance



Fig. 13 Site distance effect of DACs. (a) Net electron difference of Ni and Cu sites in different dNiCu models using sNi and sCu models as the baseline, respectively<sup>103</sup> (reproduced with permission from ref. 103 copyright 2023, John Wiley and Sons). (b) The intensity profile of the dual-atom site. (c) Distance between two adjacent metal atoms. (d) Distance distribution of two adjacent metal atoms in dual-atom sites<sup>99</sup> (reproduced with permission from ref. 99 copyright 2023, RSC).

effect is negligible. To prove this phenomenon, Chen *et al.*'s experiment showed that the site distance in DAC should be maintained within an appropriate range to ensure optimal catalytic activity.<sup>99</sup> Statistical analysis was performed on 35 randomly selected metal species, and grating diagrams and intensity spectra were utilized to illustrate the presence of dual-atom sites (Fig. 13b and c) with a bond length of 3.479 Å. Furthermore, the distribution of distances between neighboring metal atoms in the DAC was determined in Fig. 13d. The results show that the predominant interatomic distances fall within the range of 3–6 Å, and atoms with distances exceeding 9 Å are considered isolated sites and no longer exhibit site distance effects.

In addition to the distance effect between active dual metal centers, the spatial structural effect also plays a significant role in the performance of DACs.<sup>105</sup> The spatial structure effect is influenced by the position, angles, and orientations of the active sites, which determine the adsorption pattern and orientation of reactants on the catalyst surface and further influence the strength and directionality of inter-site interactions. Consequently, precise modulation of these factors could enhance the performance of DAC and fine-tune the reaction rate and selectivity. For instance, Cepitis *et al.* compared SAs with three different spatially structured DACs and revealed the importance of the geometric effect in precise material design.<sup>106</sup> Specifically, the fundamental SACs (Fig. 14a) is constrained by the OH–OOH scaling relationship,<sup>108</sup> whereas the advanced DAC model (Fig. 14b) disrupts this scaling relationship while fail to reduce

the overpotential.<sup>109</sup> Fig. 14c shows a different DAC model based on the dipyrin structure, allowing for choosing reaction pathways and switching between scaling relationships. However, replicating such a precise model in an actual electrode would demand unfeasible atomic precision.<sup>110</sup> Therefore, by shifting the research focus towards adjusting surface curvature, it was discovered that modifying surface curvature can alter the electronic structure and interatomic spacing (Fig. 14d). Four structures with different surface curvatures were established in Fig. 14e and f, and performance measurements were conducted using Co and Ni metals as models. It can be observed from Fig. 14g and h that the adsorption energy significantly decreases with increasing radius. Additionally, due to the increased number of simultaneously adsorbed intermediates, the decrease in the dissociation mechanism is more pronounced than that of the coupling mechanism (Fig. 14i and j). This study provides initial proof that surface curvature plays a crucial role in boosting catalyst performance, serving as an important direction for regulating bifunctional catalysts.

Regarding the geometric effects on DACs, optimizing the catalyst's activity, selectivity, and stability can be achieved by adjusting site distances, coordination environment, spatial arrangement, *etc.* These adjustments provide practical strategies for achieving highly efficient catalytic performances.

### 3.4 Coordination effect

The coordination environment of DACs substantially impacts coordination effects, including the number of coordinating





**Fig. 14** Spatial structural effect of DACs. Timeline of the M-N-C model development. The adsorption energies of intermediates vary with changing (a) M in the flat single-atom site model, (b) second M in the flat dual-atom site model, (c) geometry of the diporphyrin model, and (d) surface curvature of the in-pore model. Line color indicates a different M, line transparency indicates distinct geometry. (e) Top view on constructing a dual-atom site M-N-C model from an armchair nanoribbon. (f) Side view on curving the flat dual-atom M-N-C model into in-pore models. A periodic boundary condition was applied along the pore. (g) CoCo model energies, and (h) CoNi model energies. The green line indicates the ideal adsorption energy. (i) and (j) Linearized plot of free energy differences vs. the intersite distance, which is expressed as  $r^{-2}$ . Using colored points as a reference, the pore radius of the CoCo model is respectively  $\infty$ , 8.5, 8.0, 6.0, and 4.0 Å from left to right<sup>107</sup> (reproduced with permission from ref. 107 copyright 2023, ACS).

atoms at the active sites and the coordinated atoms that bonded to these sites.<sup>40,111,112</sup> For instance, Hu *et al.* investigated the influence of atomic structures of active sites on the catalytic performance of Fe-Co-N-C by comparing their configurations with different active sites (including FeCoN<sub>6</sub>, FeCoN<sub>7</sub>, and FeCoN<sub>8</sub>) for ORR and OER.<sup>113</sup> Fig. 15a illustrates the optimized configurations of the Fe-Co-N-C models. The planar structure and negative  $E_f$  of these models indicate low strain and high stability. Notably, configurations with different atomic coordination environments exhibit significant differences in  $E_f$ . This is attributed to the enhanced stability of graphene-based materials achieved by increasing the coordination number of nitrogen atoms surrounding the metal atoms.<sup>114</sup> These findings

reveal that coordination numbers play a crucial role in determining the electronic configuration of metal atoms, which subsequently impacts catalytic performance. Consequently, establishing a well-defined coordination environment for the active metal center is paramount. By comparing the  $\eta_{ORR}$  and  $\eta_{OER}$  of the active centers Fe and Co in Fe-Co-N-C, as shown in Fig. 15b and c, the Co atom exhibits superior catalytic activity for ORR and OER than the Fe atom. Therefore, the top position of the Co atom in Fe-Co-N-C serves as the primary active center for ORR and OER.

The coordination effect between dual metal active sites can also be enhanced by altering the coordinated atoms from the support material that connects with these sites to improve the



Fig. 15 The coordination effect of DACs is caused by the coordination number. (a) The configurations and  $E_f$  of Fe–Co–N–C, Fe–N–C, and Co–N–C. The gray, blue, cyan, and green spheres indicate the C, N, Co, and Fe atoms, respectively. Summary of overpotentials for (b) ORR and (c) OER on Fe–Co–N–C at the  $U = 0$  V. The corresponding overpotentials of Fe–N–C and Co–N–C are shown as the red and blue dashed lines, respectively<sup>113</sup> (reproduced with permission from ref. 113 copyright 2020, Elsevier).

catalyst's activity and stability. Recently, Huang *et al.* studied SACs and DACs with various metal centers and coordination environments to determine the most stable cobalt DAC configuration.<sup>115</sup> The structural models are shown in Fig. 16a. The results, depicted in the volcano plot of Fig. 16b, reveal the oxygen-bridged  $\text{Co}_2\text{N}_4\text{O}_2$  structure exhibits an optimal balance of activity and selectivity. PDOS analysis (Fig. 16c and d) shows the lower Co 3d orbital center ( $-2.95$  eV) in  $\text{Co}_2\text{N}_4\text{O}_2$  versus  $\text{CoN}_4$  ( $-0.98$  eV), indicating the second metal can modulate the adsorption of intermediates by shifting the d-band center. In another work, Wang and co-workers investigated the most stable coordination configuration of DACs by studying boron-coordinated  $\text{FeN}_4\text{B}$ – $\text{NiN}_4\text{B}$  models with different B and N positions.<sup>63</sup> As shown in Fig. 16e and f, the analysis revealed that boron doping induces charge transfer and asymmetric charge distribution around the central Fe/Ni atoms. This optimizes the adsorption/desorption of intermediates in ORR/OER and reduces the activation energy of the potential-determining step. In addition to investigating the most stable coordination configurations of DACs, studying how to optimize their capabilities for multistep catalytic processes is crucial. For example, Wang *et al.* studied the asymmetric coordination of metal pairs.<sup>116</sup> They synthesized atomically dispersed Fe–Se/NC and found the asymmetric Fe–Se coordination led to p–d orbital hybridization (Fig. 16g), resulting in asymmetric charge polarization. As depicted in Fig. 16h, this optimized electronic hybridization between Fe and Se improved the adsorption/desorption of intermediates. In another work, Tang *et al.*

developed distinct anionic ligands for the two metals to construct Janus dual-metal sites to maximize DAC capabilities in multistep catalysis.<sup>117</sup> They synthesized  $\text{FeCo-N}_3\text{O}_3\text{@C}$  catalysts, where Fe and Co are coordinated with N and O, respectively, acting as ORR and OER active sites (Fig. 16i). As shown in Fig. 16j, the charge density difference reveals a negative charge between the Fe and Co atoms, indicating bond formation and electron redistribution. The PDOS in Fig. 16k and l shows the higher charge density near the Fermi level for  $\text{FeCo-N}_3\text{O}_3\text{@C}$  versus single-atom references, accelerating electron transfer during the reactions.

In conclusion, the microenvironment of DACs is regulated by the interplay of four key effects. Elemental effects involve metal–metal interactions, bridging heteroatoms, and support atoms, with electronegativity differences affecting adsorption and electron transfer. Closely related to this, electronic effects arise from electron transfer, sharing, and redistribution between the metal centers, impacting the electronic structure and stability of intermediates. Additionally, geometric effects function through site distances and spatial positioning, regulating adsorption, kinetics, and configurations. Lastly, coordination effects are determined by the number and identity of coordinating atoms, stabilizing the dual-atom site and modulating the electronic structure. Carefully engineering these microenvironment aspects allows researchers to maximize the synergistic dual-atom site effect and unlock the full catalytic potential of DACs for a wide range of reactions.





**Fig. 16** The coordination effect of DACs is caused by coordination atoms. (a) Different structure models are employed in the DFT calculations. (b) Calculated activity volcano plot for a two-electron ORR. PDOS and corresponding calculated d-band centers of Co 3d for (c) CoN<sub>4</sub> and (d) Co<sub>2</sub>N<sub>4</sub>O<sub>2</sub> models<sup>115</sup> (reproduced with permission from ref. 115 copyright 2024, ACS). The differential charge density contour of (e) FeN<sub>4</sub>B–NiN<sub>4</sub>B and (f) FeN<sub>4</sub>–NiN<sub>4</sub> (ref. 63) (reproduced with permission from ref. 63 copyright 2024, John Wiley and Sons). (g) The PDOS of Fe–Se/NC models. (h) The electron localization function analysis of Fe–Se/NC<sup>116</sup> (reproduced with permission from ref. 116 copyright 2023, John Wiley and Sons). (i) The geometric structures of Janus dual-atom materials. In the ball-and-stick diagrams, the grey, blue, red, yellow, and cyan balls represent carbon, nitrogen, oxygen, iron and cobalt atoms, respectively. (j) Charge density differences for FeCo–N<sub>3</sub>O<sub>3</sub>@C. The yellow and cyan areas depict charge accumulation and depletion, respectively. The PDOS of (k) Fe 3d and (l) Co 3d for FeCo–N<sub>3</sub>O<sub>3</sub>@C, FeN<sub>4</sub>@C and CoO<sub>4</sub>@C<sup>117</sup> (reproduced with permission from ref. 117 copyright 2024, Springer Nature).

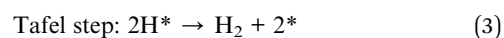
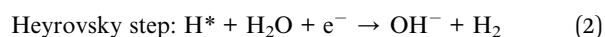
## 4. DACs for water electrolysis applications

Electrocatalytic water electrolysis has emerged as a highly efficient method for generating clean and pure hydrogen, encompassing the OER at the anode and the HER at the cathode. Considering the compatibility of the electrocatalysts in the same electrolyte for both HER and OER simultaneously, exploring highly efficient and stable bifunctional electrocatalysts toward overall water splitting is highly desirable. In recent years, DACs have shown remarkable advancements in electrochemical water electrolysis owing to their distinctive dual-atom site effects, offering a promising approach for overall water electrolysis. This chapter provides an overview of the recent progress of DACs in water electrolysis, commencing with a detailed mechanism discussion of half-reactions and followed by a thorough exploration of their performances across various pH environments.

### 4.1 DACs for hydrogen evolution reaction (HER)

HER represents the cathodic half-reaction of water electrolysis and can be facilitated by active electrocatalysts to generate pure H<sub>2</sub>.<sup>118</sup> HER typically involves a two-electron transfer process with few reaction steps,<sup>119</sup> and numerous theoretical and experimental studies have revealed different HER pathways in acidic, alkaline, and neutral conditions:

Alkaline/neutral condition:



Acidic condition:





In the above reaction steps, the  $\text{H}^*$  represents the adsorbed hydrogen. The first step is the Volmer step (1) to produce  $\text{H}^*$ , and subsequently, the reaction can proceed by the Heyrovsky step (2) and/or the Tafel step (3) to produce  $\text{H}_2$ . The adsorption energy of  $\text{H}^*$  on the active sites determines the selection of HER catalysts. Compared with acidic HER, the Volmer step in alkaline media requires proper adsorption of water molecules and additional energy to dissociate  $\text{H}_2\text{O}$ , which complicates the production of  $\text{H}^*$ . This indicates that many catalysts fail to dissociate water rapidly in the Volmer step.<sup>120</sup> Consequently, overcoming the sluggish hydrogen kinetics poses a significant challenge in alkaline HER technology.<sup>121</sup> Notably, by regulating the electronic effect between two different single atoms, heteronuclear DACs can accelerate HER better by efficiently providing dual sites for  $\text{H}^*$  and/or  $\text{OH}^*$  intermediates,<sup>105</sup> positioning them among the most effective HER electrocatalysts available.<sup>122,123</sup>

In alkaline media, Chen *et al.* designed W-W dual-atom sites on P-doped carbon materials (W-AC) through a thermal migration strategy.<sup>124</sup> The W-ACs, processing a platinum-like electronic structure, demonstrated excellent alkaline HER activity with an ultra-low overpotential of 53 mV at a current density of  $10 \text{ mA cm}^{-2}$  ( $\eta_{10}$ ) and a modest Tafel slope of  $38 \text{ mV dec}^{-1}$  (Fig. 17a), as well as good durability after 1000 CVs testing (Fig. 17b). Furthermore, by analyzing the kinetic barrier of the water dissociation step (Volmer step,  $\Delta G_{\text{H}_2\text{O}}$ ), the  $\Delta G_{\text{H}_2\text{O}}$  value of W-ACs is significantly reduced (0.24 eV), indicating a markedly accelerated Volmer step (Fig. 17c). These W-W dual-atom sites exhibit a less negative  $\text{H}^*$  adsorption energy and more favorable  $\text{OH}^*$  desorption behavior compared to those in other counterparts, thereby enhancing its alkaline HER activity. In another study, Cai *et al.* synthesized Ir/Ru dizygotic single-atom sites on CoP catalysts (Ir/Ru DSACs) and investigated their performance in alkaline HER.<sup>125</sup> As shown in Fig. 17d, IrRu DSACs exhibited a significantly higher average charge density variation than Ir and Ru SACs. It reveals a pronounced charge redistribution and the formation of a strong atomic local electric field caused by the electronic effect of IrRu DSACs. *In situ* Raman spectroscopy shows an increased stretching  $\text{H}_2\text{O}$  vibration peak on IrRu DSACs, indicating an asymmetric adsorption configuration (H-down) structure during  $\text{H}_2\text{O}$  adsorption (Fig. 17e). Consequently, IrRu DSACs facilitate interfacial water reorientation, promoting water dissociation through the electronic effects induced by a strong atomic electric field around the Ir and Ru sites, thereby enhancing alkaline HER activity (Fig. 17f).

Compared to alkaline conditions, noble metal-based DACs have been investigated in acidic HER conditions mainly due to their strong anti-acid corrosion abilities. Notably, Pt-based catalysts demonstrate outstanding performance in the HER owing to the exceptional hydrogen binding energy of Pt-H bonds, leading to reduced overpotential, elevated exchange current density, and exceptional catalytic durability.<sup>128,129</sup> Pt-based DACs are recognized as effective HER catalysts, as they

offer a viable solution to circumvent the cost limitations associated with platinum metal.<sup>130–133</sup> For example, Zhao *et al.* designed  $\text{Pt}_1\text{Ru}_1/\text{NMHCS-A}$  DAC for acidic HER. With hydrogen adsorbed on the Pt sites, this  $\text{Pt}_1\text{Ru}_1$  exhibits the lowest  $|\Delta G_{\text{H}^*}|$  value (Fig. 17g).<sup>126</sup> Additionally, the difference in charge density of the  $\text{Pt}_1\text{Ru}_1$  dimer reveals the electronic effect of Ru on Pt atoms (Fig. 17h), indicating that Pt atoms possess strong proton adsorption and reduction capabilities, thereby accelerating  $\text{H}_2$  generation and achieving optimal HER activity. As shown in Fig. 17i,  $\text{Pt}_1\text{Ru}_1/\text{NMHCS-A}$  shows high mass activity of  $3.49 \text{ A mg}^{-1}$  at  $\eta_{50}$  and an ultra-low overpotential of 22 mV at  $\eta_{10}$ , significantly surpassing those of the commercial 20 wt% Pt/C benchmark and its nanoparticle counterparts. In addition, Chao *et al.* fabricated Pd/Cu-Pt NRs consisting of Cu-Pt dual-atom sites on Pd nanoring (NRs).<sup>122</sup> The Pd/Cu-Pt NRs possess atomically dispersed ultrafine structures and exhibit superior acidic HER performance compared to Pt/C, with an overpotential of only 22.8 mV at  $\eta_{10}$  and a Tafel slope of  $25 \text{ mV dec}^{-1}$ . Da *et al.* synthesized a PtNi-NC catalyst by immobilizing PtNi on an NC substrate.<sup>134</sup> Due to the electron effects between Pt and Ni atoms, PtNi-NC DAC exhibits excellent acidic HER catalytic activity. At a current density of  $10 \text{ mA cm}^{-2}$ , it achieves an overpotential of 30 mV, comparable to commercial 20 wt% Pt/C catalysts.

Besides Pt, other noble metal-based DACs also show superior acidic HER performances. For instance, Zhou *et al.* employed a top-down approach to synthesize atomically dispersed catalysts containing Rh-Fe dual-metal sites.<sup>127</sup> As shown in Fig. 17j, the Fe/Rh-anchored NC spheres (FR-NCS) demonstrated outstanding catalytic activity, exhibiting a positive shift in overpotential of 22 mV at  $\eta_{10}$  compared to commercial Pt/C catalysts. Fig. 17k and l illustrate the charge distribution to highlight the electron redistribution effect of Fe on Rh. When hydrogen is adsorbed, the electrons tend to cluster around the atomically dispersed Fe atoms (dark cyan region), while Rh atoms share electrons with the hydrogen species (red region). This charge redistribution phenomenon indicates that the Fe atoms in FR-NCS catalysts facilitate efficient charge distribution, while the Rh sites favor hydrogen adsorption and evolution. Consequently, the electronic effect between the dual-atom sites contributes to the exceptional acidic HER performance.

Considering the universality and versatility of DACs, it is crucial to develop DACs suitable for all pH HER. Employing DFT calculations to identify DACs with excellent HER performance under all pH conditions is a viable strategy. Recently, Jiao *et al.* systematically screened a series of highly active Co-M-DAC ( $\text{M} = \text{Fe}, \text{Cu}, \text{Zn}$ ) DACs based on DFT calculations and experimental testing (Fig. 18a).<sup>86</sup> The calculated  $\Delta G_{\text{H}^*}$  values for hydrogen adsorption on these Co-M DACs were calculated, as shown in Fig. 18b. A  $\Delta G_{\text{H}^*}$  value close to 0 eV indicates stronger hydrogen evolution capability, signifying higher HER performance. The CoCu DAC@CP exhibited enhanced HER activity ( $90 \text{ mV}$  at  $\eta_{20}$ ) under 0.5 M  $\text{H}_2\text{SO}_4$  conditions with a Tafel slope of  $92.4 \text{ mV dec}^{-1}$ , suggesting a Volmer-Heyrovsky mechanism. In contrast, Co SAC@CP exhibited a more significant Tafel slope of  $136.7 \text{ mV dec}^{-1}$ , indicating a Volmer mechanism (Fig. 18c and d). In contrast, under 1.0 M KOH, CoFe DAC@CP demonstrated





**Fig. 17** The HER performance of DACs in alkaline or acidic conditions. (a) Polarization linear sweep voltammetry (LSV) curves for W-ACs and reference catalysts in 1.0 M KOH solution. (b) LSV curves for W-SAs and W-ACs before and after 1000 CVs. Inset: the chronopotentiometry measurements at a current density of 10 mA cm<sup>-2</sup>. (c) The calculated free energy diagrams for water dissociation steps on W-ACs and the references<sup>124</sup> (reproduced with permission from ref. 124 copyright 2022, Springer Nature). (d) Total electric field and charge variations of Ir–Ru pairs in IrRu DSACs. (e) The *in situ* Raman spectra of IrRu DSACs. (f) Electrochemical polarization curves of the as-prepared catalysts<sup>125</sup> (reproduced with permission from ref. 125 copyright 2023, John Wiley and Sons). (g) Calculated  $\Delta G_{H^*}$  values of different structures. (h) The charge density difference before and after adsorption of H on the Pt atom in the Pt<sub>1</sub>Ru<sub>1</sub> dimer. (i) Mass activity and  $\eta_{10}$  of all samples tested in 0.5 M H<sub>2</sub>SO<sub>4</sub> solution<sup>126</sup> (reproduced with permission from ref. 126 copyright 2022, ACS). (j) Polarization curves of FR-NCS and reference samples. PDOS (k) before and (l) after H being adsorbed on the Rh atoms of FR-NCS with the corresponding charge distribution changes in the upper panels of each PDOS. Atomically dispersed Fe via the Rh–Fe bond retrieves some electrons in Rh to make the suitable number of active electrons in the Rh active site, facilitating a moderate value of  $\Delta G_{H^*}$ . The red, blue, and green areas correspond to the electron distributions in Rh, Rh–Fe bonding, and Fe atoms, respectively<sup>127</sup> (reproduced with permission from ref. 127 copyright 2020, John Wiley and Sons).

superior HER activity (21.5 mV@ $\eta_{20}$ ) compared to other catalysts and exhibited a smaller Tafel slope of 62.9 mV dec<sup>-1</sup> (Fig. 18e and f). This approach enables the pre-screening of relevant DACs, aiding in selecting optimal catalysts.

Notably, several studies have confirmed that Mo-based materials exhibit excellent HER performance across all pH ranges. For instance, Yang *et al.* developed a novel O-

coordinated W–Mo heterodimer (W<sub>1</sub>Mo<sub>1</sub>–NG) supported on nitrogen-doped graphene.<sup>135</sup> This heterodimer possesses a unique W–O–Mo–O–C structure that influences the electron transfer between the catalyst and adsorbates (the inset in Fig. 18g), resulting in high HER performance of W<sub>1</sub>Mo<sub>1</sub>–NG under both acidic and alkaline conditions. As shown in Fig. 18g and h, W<sub>1</sub>Mo<sub>1</sub>–NG exhibits a significantly lower overpotential of



**Fig. 18** The HER performance of DACs in all pH conditions. (a) Structure diagram of CoM DACs, in which gray is C, dark blue is N, light blue is Co, and white is H; M = Cu, V, W, Mo, Fe, Zn, and Co. (b)  $\Delta G_{H^+}$  on single Co atoms catalyst and different CoM DACs. (c) Polarization curves and (d) the corresponding Tafel plots of various catalysts in 0.5 M  $H_2SO_4$ . (e) Polarization curves and (f) the corresponding Tafel plots of different catalysts in 1.0 M KOH<sup>86</sup> (reproduced with permission from ref. 86 copyright 2023, Elsevier). Polarization curves and the corresponding Tafel plots of  $W_1Mo_1$ -NG and reference samples in (g and h) 0.5 M  $H_2SO_4$  and (i and j) 1.0 M KOH, respectively. Inset in (g) optimized geometries and possible active sites for H adsorption on  $W_1Mo_1$ -NG systems. Atom colors: cyan, Mo; royal blue, W; dark gray, C; blue, N; red, O; orange, H. (k) Comparison of the  $\eta_{10}$  in 0.5 M  $H_2SO_4$  and 1.0 M KOH solutions for  $W_1Mo_1$ -NG and reported W- and Mo-based HER electrocatalysts<sup>135</sup> (reproduced with permission from ref. 135 copyright 2020, AAAS). (l) The overpotential of FeMo@CoNi-OH/ $Ni_3S_2$  at 10 mA  $cm^{-2}$  compared to other typical catalysts with HER properties in a wide pH range. (m) Chronopotentiometry measurements of FeMo@CoNi-OH/ $Ni_3S_2$  in 1.0 M KOH, 1.0 M PBS, and 0.5 M  $H_2SO_4$  solution<sup>136</sup> (reproduced with permission from ref. 136 copyright 2023, Elsevier).

24 mV@ $\eta_{10}$  and a noticeably low Tafel slope of 30 mV  $dec^{-1}$  in acidic electrolytes. Furthermore,  $W_1Mo_1$ -NG exhibits outstanding HER performance ( $\eta_{10}$  = 67 mV) in alkaline electrolytes (Fig. 18i and j). Comparative analysis of overpotential values, as shown in Fig. 18k, reveals that the performance of  $W_1Mo_1$ -NG surpasses the reported Mo- and W-based catalysts under both acidic and alkaline conditions. In another work, Fang *et al.* synthesized rod-shaped  $Ni_3S_2$  materials modified with single-atom Fe-Mo and encapsulated in nickel-based nanosheets (FeMo@CoNi-OH/ $Ni_3S_2$ ).<sup>136</sup> The introduction of Fe and Mo altered the electronic structure of the catalyst, leading to enhanced conductivity and improved HER performance. Under alkaline, acidic, and neutral conditions, FeMo@CoNi-OH/ $Ni_3S_2$  exhibited excellent HER performance and remarkable

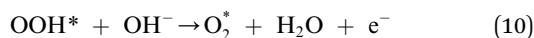
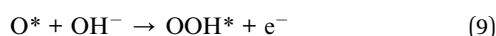
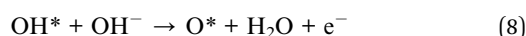
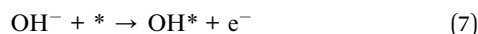
stability over 30 hours, outperforming the majority of previously studied electrocatalysts (Fig. 18l and m). The modulation of electronic effects within the FeMo@CoNi-OH/ $Ni_3S_2$  microenvironment results in significant electron redistribution, leading to a remarkable enhancement in catalytic performance. Shi *et al.* also developed a nitrogen-coordinated Ir-Mo DAC on a carbon matrix (Ir-Mo DAC/NC).<sup>137</sup> The Ir-Mo DAC/NC demonstrated exceptional HER performance, with a significantly reduced HER overpotential of 11.3 mV and 23 mV@ $\eta_{10}$  in acidic and alkaline conditions, respectively. DFT calculations confirmed that the presence of Ir-Mo pair sites significantly enhanced the HER kinetics by facilitating the accumulation of electron density on the Ir-Mo sites.



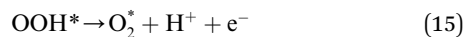
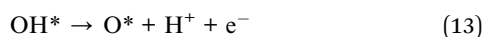
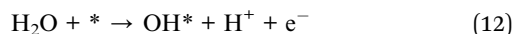
## 4.2 DACs for oxygen evolution reaction (OER)

The OER is a kinetically sluggish process involving a four-electron and proton-coupled reaction, *i.e.*,  $2\text{H}_2\text{O} \rightarrow \text{O}_2 + 4\text{H}^+ + 4\text{e}^-$ , which is the oxidation process and serves as the anodic half-reaction in water electrolysis. Therefore, the OER is the critical process that governs the overall efficiency of electrochemical water electrolysis.<sup>138–140</sup> The reaction steps in both acidic and alkaline media are as follows:

Alkaline conditions:



Acidic conditions:



The OER mechanism in both alkaline and acidic electrolytes includes a four steps process (outlined in formulae (7) to (16)), each step coupled with an electron transfer, and involves the formation of multiple intermediates ( $\text{OH}^*$ ,  $\text{O}^*$ ,  $\text{OOH}^*$ , and  $\text{O}_2^*$ ) that ultimately lead to the production of oxygen gas ( $\text{O}_2$ ).<sup>141</sup> Research on its reaction mechanism allows for the theoretical design of more efficient catalyst materials. Three distinct OER mechanisms have been identified: Adsorbate Evolution Mechanism (AEM), Lattice Oxygen Mechanism (LOM), and Oxide Path Mechanism (OPM).<sup>141–143</sup> The AEM pathway is widely accepted in transition metal-based OER electrocatalysts, which involves four electron-proton transfer steps accompanied by metal oxidation and reduction. However, AEM is limited in explaining the origins of catalytic activity and, thereby, the mechanism observed in many highly active acidic OER catalysts. Alternatively, for LOM, lattice oxygen participates in the formation of  $\text{O}_2$  through direct O–O coupling, thereby surpassing the theoretical overpotential limit of the AEM.<sup>144</sup> However, LOM leads to the formation of oxygen vacancies, accelerating the dissolution of metals and resulting in rapid catalyst deactivation.<sup>145</sup> This phenomenon significantly affects the electrochemical stability of the catalyst. Recently, the OPM pathway, involving dual catalytic sites, facilitates direct coupling of O–O bond and the formation of  $\text{O}_2$  without the participation of lattice oxygen has recently gained significant

attention.<sup>89,143,146</sup> Studies have shown that OPM is typically more advantageous in dual-site systems, such as DACs.<sup>144,147</sup> The differing binding strengths of intermediates on the two sites can minimize the maximum free energy. In contrast, the charge difference between the site atoms facilitates the formation of  $^*\text{O}-\text{O}^*$  coupling, ultimately leading to the generation of  $\text{O}_2$ . These phenomena drive the transition of the OER pathway towards dual-site OPM, synergistically reducing the overpotential of OER and effectively enhancing its activity.

To elucidate the OER mechanism of optimal DACs, Fang *et al.* synthesized graphene-based heterogeneous DACs ( $\text{M}'\text{M}@\text{NCs}$ ) as model catalysts to explore their OER activities and related  $^*\text{O}-\text{O}^*$  coupling mechanism at dual-metal sites (OPM pathway).<sup>89</sup> A systematic exploration of 86 DAC systems ( $\text{FeM}$ ,  $\text{CoM}$ ,  $\text{NiM}$ , and  $\text{CuM}$ ) along a preselected O–O coupling pathway, as shown in Fig. 19a,  $\text{NiPd}@\text{NC}$  (0.06 V),  $\text{CuPd}@\text{NC}$  (0.07 V), and  $\text{CuPt}@\text{NC}$  (0.08 V) outperform the conventional overpotential bottleneck of 0.37 V in AEM pathway. The preferred  $^*\text{OH}$  adsorption on  $\text{M}_2$  dimers avoids the scaling issues between  $^*\text{OOH}$  and  $\text{OH}$ , bringing  $\eta$  closer to 0 V. More importantly, by comparing the Gibbs free energy barriers of two competing reaction pathways ( $^*\text{OH} \rightarrow ^*\text{OH}-^*\text{OH}$  vs.  $^*\text{OH} \rightarrow ^*\text{O}$ ), the potential-determining step (PDS) for DACs primarily occurs through the coupling of  $^*\text{O}-\text{O}^*$  to generate  $^*\text{O}_2$  (Fig. 19b–d). They further demonstrated that the synergistic effect between  $\text{M}_2$  dimers of selected  $\text{M}'\text{M}@\text{NCs}$  modifies the d-band center and M–O bond states, facilitating the moderate adsorption of  $^*\text{OH}$  and enhancing OER activity *via* the  $^*\text{O}-\text{O}^*$  coupling mechanism. Moreover, the constant potential method (CPM) calculations reveal that OER activity on the diatomic catalysts is pH-dependent, suggesting that three selected DACs, *i.e.*,  $\text{NiPd}@\text{NC}$ ,  $\text{CuPd}@\text{NC}$ , and  $\text{CuPt}@\text{NC}$ , are effective under both acidic and alkaline conditions (Fig. 19e–g). Therefore, the  $^*\text{O}-\text{O}^*$  coupling mechanism is a crucial pathway for future OER research, presenting an alternative approach distinct from the traditional AEM pathway.

Alkaline water electrolyzers, such as anion exchange membrane (AEM) electrolyzers, have been widely applied in industrial  $\text{H}_2$  production due to their relative simplicity and cost-effectiveness compared to other electrolyzers. In recent years, significant progress has been made in developing DACs that exhibit excellent OER activity and stability under alkaline conditions.<sup>55,93,119,148</sup> Among them, Co-, Ni- and Fe-based catalysts have gained wide attention due to their excellent performance in OER.<sup>78,149</sup> This is attributed to their ability to modulate the 3d electron configuration and spin state, the diversity of their crystal and electronic structures, as well as their natural abundance, making them promising alternatives to precious metal catalysts.<sup>150</sup> Notably, Co- and Fe-based materials are widely utilized as OER electrocatalysts.<sup>151</sup> For example, Zhang *et al.* synthesized Fe–Co DACs with both steric and planar structures, finding that the planar Fe–Co DACs show exceptional alkaline OER performance, with an overpotential of 190 mV@ $\eta_{10}$ , a low Tafel slope of 33.6 mV  $\text{dec}^{-1}$  (Fig. 20a and b), and stability of exceeding 160 h.<sup>144</sup> The planar Fe–Co dual-sites optimize the Co atom's electronic structure and electron transfer rates, thereby enhancing catalytic efficiency and

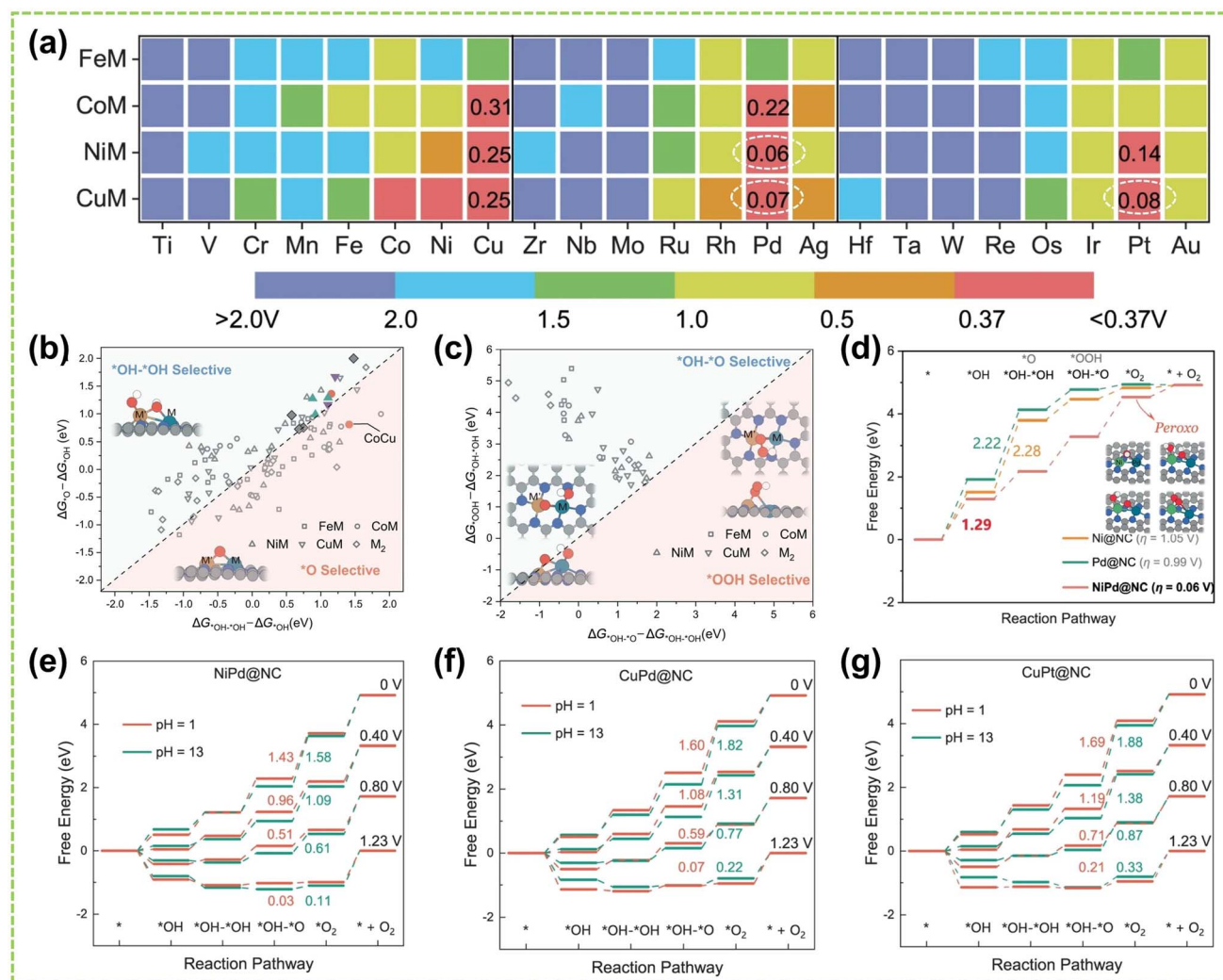


Fig. 19 The OER performance of DACs in all pH conditions. (a) Activity map. The computed overpotential of the FeM, CoM, NiM, and CuM dimers on M'M@NC. (b)  $\Delta G$  of  $^*OH$  toward  $^*OH-^*OH$  versus  $^*OH \rightarrow ^*O + H^+ + e^-$ . Data points screened for high-performance homo- and hetero-nuclear catalysts are colored. (c)  $\Delta G$  of  $^*OH-^*OH \rightarrow ^*OH-^*O + H^+ + e^-$  versus  $^*OH-^*OH \rightarrow ^*OOH + H^+ + e^-$ . Insets: the schematic configuration of the different intermediates adsorbed on M'M@NC. The white, red, gray, and blue spheres in the atomic models represent H, O, C, and N atoms, respectively. (d) Free energy diagrams of NiPd@NC and the respective corresponding SACs for OER. Inset: the optimized configurations of oxygenated intermediates represented as NiPd@NC. Free energy diagrams of (e) NiPd@NC, (f) CuPd@NC, and (g) CuPt@NC at different potentials with pH = 1 and 13 and the corresponding  $\Delta G_{^*OH-^*O \rightarrow ^*O_2}$  are labeled<sup>89</sup> (reproduced with permission from ref. 89 copyright 2023, Springer Nature).

highlighting the significance of microenvironmental regulation and its impact on the geometric effect in DACs. Surface-enhanced infrared spectroscopy was employed to identify the  $^*O-O^*$  bond that existed in both the planar-Fe-Co DAC and steric-Fe-Co DACs (Fig. 20c and d). As the voltage increased, peaks at  $1470\text{ cm}^{-1}$  and  $1468\text{ cm}^{-1}$  gradually appeared, corresponding to the  $^*O-O^*$  intermediates, intermediates preceding the release of  $O_2$ . This observation confirms that the catalysts follow the OPM pathway during the OER process in an alkaline solution. In another work, Liu *et al.* synthesized adjacent Fe-Co DACs embedded in nitrogen-doped graphite carbon (FeCo-DACs/NC).<sup>152</sup> As shown in Fig. 20e and f, FeCo-DACs/NC exhibited outstanding alkaline OER performance, achieving a low overpotential of  $370\text{ mV}@ \eta_{10}$  and a Tafel slope of  $82.7\text{ mV dec}^{-1}$ . They also investigated the distance effects in FeCo DACs

by varying distances between two metal sites. By analyzing the free energy profiles and density of states (DOS) during the reaction process (Fig. 20g and h), the FeCo1 with the shortest metal distance exhibits optimal performance due to the lowest reaction-free energy and smallest band gap energy. This indicates that the enhanced electron conductivity in FeCo-DACs/NC not only facilitated electron transfer but also, through the synergistic interaction between the two metals, significantly impacted the rate-determining step (RDS) and reaction barriers for OER, thereby boosting the electrocatalytic process. Besides, He and co-workers also reported the Fe-Co dual-metal site catalyst through effective chemical doping and controlled thermal decomposition of ZIF-8, processing outstanding alkaline OER catalytic activity and stability.<sup>154</sup>



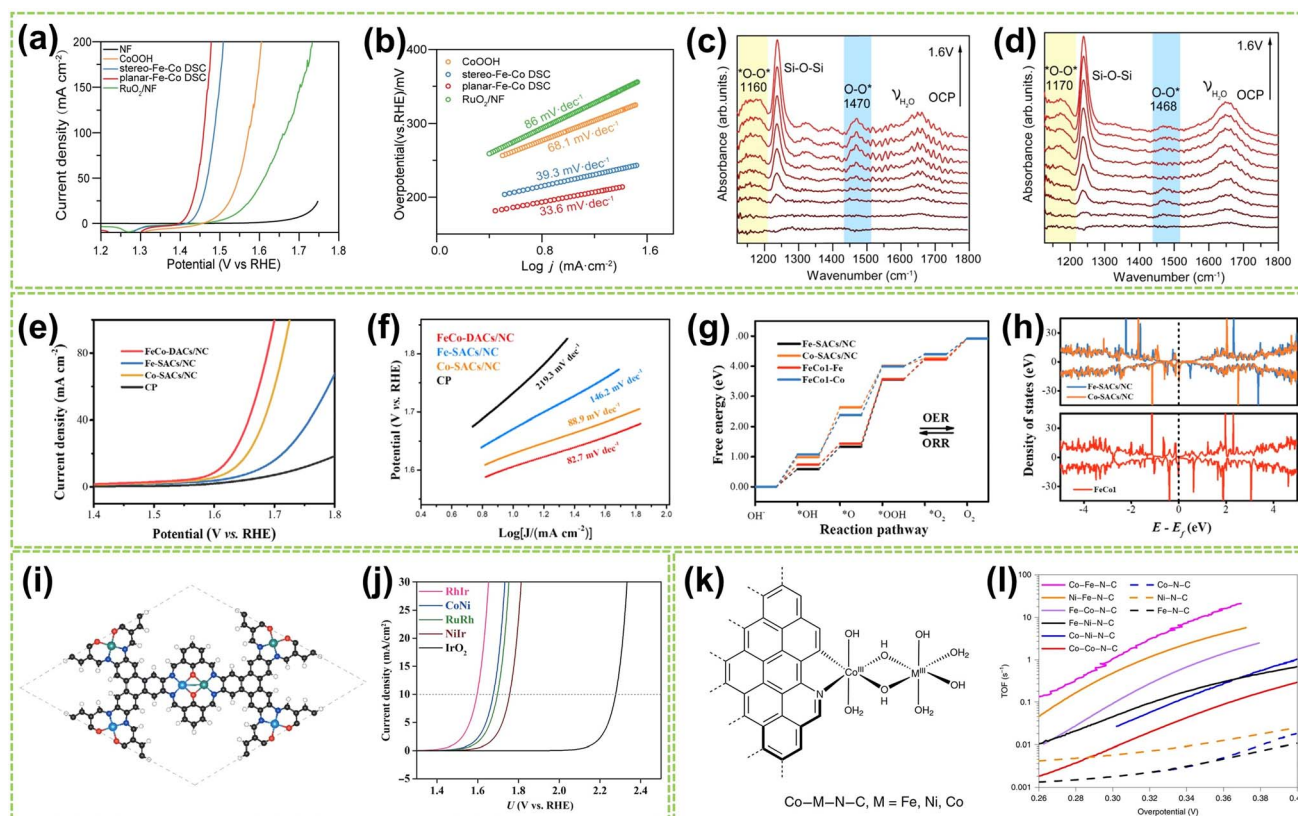


Fig. 20 The OER performance of DACs in alkaline conditions. (a) Polarization curves and (b) Tafel slopes of stereo-Fe-Co DAC, planar-Fe-Co DAC, and reference samples. The measurements were conducted in 1.0 M KOH. *In situ* FTIR spectra were recorded in the potential range of OCP to 1.6 V vs. RHE for (c) planar-Fe-Co DAC and (d) stereo-Fe-Co DAC<sup>144</sup> (reproduced with permission from ref. 144 copyright 2024, *Proceedings of the National Academy of Sciences*). (e) LSV curves and (f) Tafel plots of FeCo-DACs/NC and references for OER. (g) Mechanism investigation of free energy diagrams for OER and ORR. (h) DOS patterns of FeCo1, Co-SACs/NC, and Fe-SACs/NC<sup>152</sup> (reproduced with permission from ref. 152 copyright 2022, John Wiley and Sons). (i) Optimized geometric configurations of TM<sub>1</sub>TM<sub>2</sub>-COF. (j) Simulated polarization curves for RhIr-COF, CoNi-COF, RuRh-COF, and NiIr-COF in comparison with that of IrO<sub>2</sub>(110)<sup>153</sup> (reproduced with permission from ref. 153 copyright 2022, Springer Nature). (k) The structure of Co-based single-atom precatalysts. (l) The activity of single-atom catalysts was obtained from the first LSV curve of these catalysts in Fe-free 1 M KOH<sup>78</sup> (reproduced with permission from ref. 78 copyright 2021, Springer Nature).

Except for Fe-Co DACs, some other DACs have outstanding alkaline OER performance. For instance, Wan and co-workers developed a dual-site NiFe catalyst on N-doped graphene nanosheets, achieving an overpotential of 270 mV at 10 mA cm<sup>-2</sup> in the alkaline electrolyte.<sup>55</sup> Similarly, atomically dispersed Ni/Co dual active sites anchored on NC hollow prisms (a-NiCo/NC) exhibit excellent alkaline OER performance.<sup>155</sup> Zhou *et al.* constructed various homonuclear DACs (2TM-COF, which TM represents transition metal) and heteronuclear DACs (TM<sub>1</sub>TM<sub>2</sub>-COF) using two-dimensional (2D) COF materials as the substrate.<sup>153</sup> As depicted in Fig. 20i, the stable geometry of the TM<sub>1</sub>TM<sub>2</sub>-COF features a metal dimer, where each TM atom is surrounded by two N atoms and two O atoms. Importantly, as shown in Fig. 20j, the lower OER starting potentials of RhIr-COF and CoNi-COF at 10 mA cm<sup>-2</sup> further demonstrate the promise of these DACs for OER applications. Similarly, Bai and co-workers synthesized six N-doped carbon-supported DACs containing Co, Fe, or Ni.<sup>78</sup> The structure of Co-based single-atom precatalysts is shown in Fig. 20k. These DACs were experimentally confirmed to exhibit excellent OER

performance under alkaline conditions, with the Co-Fe-N-C and Ni-Fe-N-C catalysts achieving TOF values at the high end among known OER catalysts (Fig. 20l). This superior catalytic activity is attributed to the synergistic effect of the metal ions and their coordination environment, which facilitates nucleophilic attack on the M(IV)=O species and promotes the formation of O-O bonds.

Notably, the durability of OER catalyzed by DACs is a critical concern, especially when they are supported by carbon-based materials. Carbon-based supports are widely used due to their favorable surface structures for dispersed metal sites and excellent electron transport. However, carbon materials are susceptible to oxidative corrosion under OER conditions, particularly in acidic environments, leading to low catalyst stability.<sup>156,157</sup> Besides, oxidative corrosion of the carbon support can cause the loaded active species to become detached or undergo structural collapse, leading to reduced catalyst stability.<sup>158</sup> Strategies to address the durability challenge include nitrogen doping in the carbon support, using multilayer graphene, or exploring alternative corrosion-resistant and

conductive supports to enhance the durability of atomically dispersed active sites under demanding OER conditions.<sup>46,136,159</sup> A typical example is a series of methods proposed by Zoller *et al.* to increase the durability of carbon-based OER catalysts by using three-dimensional porous and multi-doped graphene-like structures as the best nanoform or by combining carbon with transition metal/oxide nanoparticles to produce synergistic effects.<sup>159</sup>

### 4.3 DACs for the overall water electrolysis

The overall reaction during electrochemical water electrolysis in an electrolytic cell is as follows:



As discussed in the above sections on HER and OER,  $\text{H}_2$  is generated at the cathode and  $\text{O}_2$  at the anode during water electrolysis. Compared to single-function electrocatalysts, developing bifunctional catalysts that are efficient for both HER and OER presents significant challenges but also holds substantial potential for practical applications. DACs, with their unique dual-atom sites and associated site effects, are promising candidates for effective overall water electrolysis. This section reviews current strategies to boost the performance of bifunctional DACs in both HER and OER to improve their overall effectiveness in practical electrolysis applications.

Zhang and co-workers successfully synthesized a Pt@Ni-FeCo-E DAC with two types of Pt dual-atom structures, a 2Pt dimer with 2.6 Å interatomic distance and a Pt<sub>2</sub> pair with 0.9 Å distance.<sup>32</sup> The nickel-based hydroxide substrate was enriched



Fig. 21 The overall water electrolysis performance of DACs in alkaline or acidic conditions. LSV curves toward (a) OER, (b) HER, and (c) overall water-electrolysis in 1.0 M KOH for all investigated and commercial (Pt/C or RuO<sub>2</sub>) catalysts. (d) Stability test for overall water electrolysis cells<sup>40</sup> (reproduced with permission from ref. 40 copyright 2023, Springer Nature). (e) OER and (f) HER polarization curves of FeMo@CoNi-OH/Ni<sub>3</sub>S<sub>2</sub> and reference samples with a scan rate of 5 mV s<sup>-1</sup> in 1.0 M KOH solution. (g)  $\Delta G_{\text{H}^*}$  on Fe'/CoNi-OH/Ni<sub>3</sub>S<sub>2</sub>, Mo'/CoNi-OH/Ni<sub>3</sub>S<sub>2</sub>, Fe'Mo@CoNi-OH/Ni<sub>3</sub>S<sub>2</sub> and FeMo@CoNi-OH/Ni<sub>3</sub>S<sub>2</sub> reaction surfaces. (h)  $\Delta G$  diagram on the Fe'Mo@CoNi-OH/Ni<sub>3</sub>S<sub>2</sub> (ref. 136) (reproduced with permission from ref. 136 copyright 2023, Elsevier). Polarization curves of commercial Pt/C, Ir NCs, and IrNi NCs in 0.5 M H<sub>2</sub>SO<sub>4</sub> for (i) OER and (j) HER. (k) Polarization curve for overall water electrolysis of IrNi NCs in a two-electrode configuration and the voltage difference between HER and OER. (Inset: polarization curves of IrNi NCs for HER and OER). All data are recorded at a scan rate of 5 mV s<sup>-1</sup> without *iR*-compensation<sup>160</sup> (reproduced with permission from ref. 160 copyright 2017, John Wiley and Sons). (l) Polarization curves and (m) Tafel plots of interlayer-confined NiFe@MoS<sub>2</sub>, planar NiFe@MoS<sub>2</sub>, interlayer-confined Sv@MoS<sub>2</sub>, and IrO<sub>2</sub> for OER. (n) Polarization curves and (o) Tafel plots of interlayer-confined NiFe@MoS<sub>2</sub>, planar NiFe@MoS<sub>2</sub>, interlayer-confined Sv@MoS<sub>2</sub>, bare glassy carbon electrodes and 20% Pt/C for HER<sup>161</sup> (reproduced with permission from ref. 161 copyright 2023, John Wiley and Sons).



with cationic vacancies, which served as anchoring sites for the Pt dual-atoms. The 2Pt dimer and Pt<sub>2</sub> pair structures provided HER and OER bifunctional catalytic activity, respectively, resulting in excellent overall water splitting performance under alkaline conditions (14 mV@ $\eta_{10}$  for HER, 234 mV@100 mA cm<sup>-2</sup> for OER, and 1719/1755 mV@100 mA cm<sup>-2</sup> for overall water splitting in freshwater/seawater). This surpassed commercial Pt/C and most reported Pt-based catalysts. The catalyst also demonstrated robust long-term HER and OER stability over 96 hours. In another work, Wang *et al.* successfully generated Co<sub>2</sub>N<sub>5</sub> dual atoms (Co<sub>2</sub>-N-HCS-900), exhibiting excellent alkaline OER (Fig. 21a) and HER (Fig. 21b) activities.<sup>40</sup> The overpotentials achieved by the Co<sub>2</sub>-N-HCS-900-based system at current densities of 10, 100, and 200 mA cm<sup>-2</sup> are competitive with commercial Pt/C + RuO<sub>2</sub> (Fig. 21c). Besides, as shown in Fig. 21d, the Co<sub>2</sub>-N-HCS-900 potential remains constant after continuous operation for 1000 hours in 1.0 M KOH, demonstrating its exceptional stability. These results reveal the significant role of neighboring Co atoms in achieving enhanced bifunctional activity in the Co<sub>2</sub>N<sub>5</sub> structure. In addition, due to the high intrinsic HER and OER activities of Mo-SACs<sup>162</sup> and Fe-SACs,<sup>51,163</sup> Chen *et al.* developed a Fe and Mo dual-site single-atom catalyst supported by CoNi-OH/Ni<sub>3</sub>S<sub>2</sub> heterojunction, referred to as FeMo@CoNi-OH/Ni<sub>3</sub>S<sub>2</sub>.<sup>136</sup> This bifunctional DAC exhibited outstanding OER (160 mV) and HER (89 mV) activities (Fig. 21e and f) and stability in alkaline conditions. When Mo serves as the active site,  $\Delta G_{H^*}$  is negative and relatively small, indicating easy adsorption of H\* and confirming Mo as the primary active site in the HER process (Fig. 21g). In addition, the  $\Delta G_{H^*}$  approaches zero with the coexistence of Fe and Mo, suggesting that the synergistic effect

between Fe and Mo enhances the HER performance. For OER, as shown in Fig. 21h, the reaction step of O\* conversion to OOH\* serves as the RDS and the  $\Delta G$  for RDS (0.81 eV) is significantly lower than the  $\Delta G$  for RDS of other reference catalysts, indicating that the site effect of Fe and Mo also enhances the OER performance. Recently, Zhao *et al.* employed a surface modification strategy to construct a Co/Ni DAC (CoNi-Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub>) on MXene substrates.<sup>53</sup> CoNi-Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> electrocatalysts exhibited excellent water-electrolysis performance in alkaline media, with overpotentials of 31 mV for HER and 241 mV for OER at 10 mA cm<sup>-2</sup>. Furthermore, CoNi-Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> electrocatalysts demonstrated high operational stability exceeding 100 hours at industrially relevant current densities of 500 mA cm<sup>-2</sup> for both OER and HER. They use DFT calculations to reveal the d-band centers of CoNi-Ti<sub>3</sub>C<sub>2</sub>T<sub>x</sub> are modulated by Co and Ni atoms, providing active adsorption sites and significantly enhancing the intrinsic activity of the electrocatalyst. In addition, the atomically dispersed Ru/Ni dual-atom sites, achieved by decorating Ru on a nickel vanadium layered double hydroxide (LDH) support, also exhibit excellent OER and HER performance under alkaline conditions.<sup>164</sup>

Acidic conditions offer a higher proton concentration and accelerate the water electrolysis process, making it crucial to develop highly efficient catalysts that can sustain both high catalytic activity and stability under such conditions. Since iridium oxide-based catalysts are the only commercial material that has demonstrated activity and stability for OER during long-term testing, this discussion particularly focuses on Ir-based DACs for acidic overall water electrolysis.<sup>160,165,166</sup> For example, Pi *et al.* incorporated Ir as one of the metals in dual-atom sites to construct a series of carbon-supported, highly

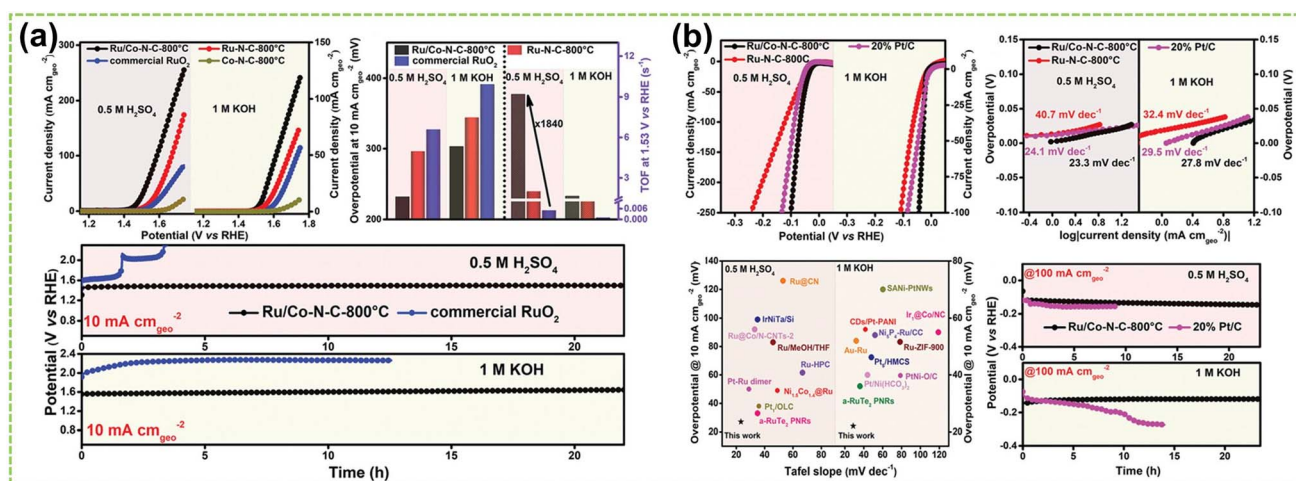


Fig. 22 The overall water electrolysis performance of DACs in wide-pH conditions. (a) Electrocatalytic OER performance of Ru/Co-N-C-800 °C compared with other samples. Above left: OER polarization curves were obtained at a scan rate of 5 mV s<sup>-1</sup> in N<sub>2</sub>-saturated 0.5 M H<sub>2</sub>SO<sub>4</sub> and 1 M KOH. Above right: comparison of overpotential to achieve a current density of 10 mA cm<sup>-2</sup> and TOF value at 1.53 V versus RHE in 0.5 M H<sub>2</sub>SO<sub>4</sub> and 1 M KOH. Below: the chronopotentiometry of Ru/Co-N-C-800 °C and commercial RuO<sub>2</sub> in 0.5 M H<sub>2</sub>SO<sub>4</sub> and 1 M KOH. (b) Electrocatalytic HER performance of Ru/Co-N-C-800 °C compared with other samples. Above left: HER polarization curves obtained at a scan rate of 5 mV s<sup>-1</sup> in N<sub>2</sub>-saturated 0.5 M H<sub>2</sub>SO<sub>4</sub> and 1 M KOH. Above right: the corresponding Tafel plots in 0.5 M H<sub>2</sub>SO<sub>4</sub> and 1 M KOH. Below left: comparison of overpotential at 10 mA cm<sup>-2</sup> and Tafel slope with previously reported HER catalysts in acid and alkaline conditions. Below right: the chronopotentiometry curves of Ru/Co-N-C-800 °C and commercial Pt/C in 0.5 M H<sub>2</sub>SO<sub>4</sub> and 1 M KOH<sup>169</sup> (reproduced with permission from ref. 169 copyright 2022, John Wiley and Sons).

dispersed IrM (M = Ni, Co, Fe) nanoclusters (NCs) electrocatalysts.<sup>160</sup> The polarization curves of the catalysts in 0.5 M H<sub>2</sub>SO<sub>4</sub> indicated that IrNi NCs exhibited the best catalytic activity for both OER (Fig. 21i) and HER (Fig. 21j). Consequently, IrNi nanomaterials were employed as both cathode and anode catalysts in a dual-electrode water electrolysis system, with the polarization curves shown in Fig. 21k. The voltage difference ( $\Delta V$ ) between the two half-reactions (HER and OER) was calculated based on the principle of charge conservation for comparison, demonstrating alignment with the overall water-electrolysis polarization curve of IrNi NCs and indicating a steady-state process. Recent findings suggest that confining metal dual-atoms within the van der Waals gaps of two-dimensional layered materials can significantly improve the kinetics and reduce the energy requirements of catalytic reactions.<sup>167,168</sup> Jiang *et al.* synthesized a unique type of NiFe@MoS<sub>2</sub> DACs by assembling Ni and Fe dual-atoms into the interlayer of MoS<sub>2</sub> nanosheets.<sup>161</sup> This distinctive interlayer confinement structure protects the metal dual-atoms, preventing their aggregation and deactivation during prolonged reactions in harsh acidic conditions, thereby sustaining the catalyst's activity over time. Fig. 21l–o evaluates the electrochemical performance of the catalyst in a 0.5 M H<sub>2</sub>SO<sub>4</sub> solution and confirms that the interlayer-enclosed NiFe@MoS<sub>2</sub> exhibits higher catalytic activity for HER and OER compared to unconstrained NiFe@MoS<sub>2</sub>, as evidenced by the Tafel slopes and overpotentials. Due to its simple synthesis method and wide applicability, this unique and effective structure is expected to advance the research on DAC synthesis under acidic conditions.

In another work, Rong *et al.* developed an atomically dispersed Ru–Co dual-site catalyst that modulated the affinity of intermediates on the Ru surface, enabling a highly active bifunctional catalyst for water splitting under both acidic and basic conditions.<sup>169</sup> As shown in Fig. 22g, the Ru/Co–N–C-800 °C catalyst exhibited low OER overpotentials of 232 and 276 mV in 0.5 M H<sub>2</sub>SO<sub>4</sub> and 1 M KOH, respectively, along with a high TOF of 9.20 s<sup>−1</sup> at 1.53 V and excellent durability over 20 hours of operation in both environments. Furthermore, the catalyst demonstrated superior HER activity, requiring only 17 mV and 19 mV overpotential to deliver 10 mA cm<sup>−2</sup> in 0.5 M H<sub>2</sub>SO<sub>4</sub> and 1 M KOH, respectively, even outperforming commercial Pt/C (Fig. 22h).

## 5. Summary and perspective

Developing highly efficient catalysts for electrochemical water electrolysis has been a pivotal research aspect of producing high-purity hydrogen. In recent years, DACs, featured with two specifically active metal centers arranged in a specific configuration on support, have garnered significant attention as efficient catalysts towards HER, OER, and overall water electrolysis. Notably, DACs provide dual-atom sites that overcome the shortages of SACs, such as limited adsorption modes and relatively low reaction activity, while retaining the advantages of SACs, including unsaturated coordination structures, high metal utilization, and relatively homogeneous active sites. Microenvironment regulation plays a vital role in fully utilizing

and maximizing the dual-atom site effects in DACs. In this review, we first classified DACs into three categories, *i.e.*, homonuclear, heteronuclear, and molecular configurations, based on their different dual-atom configurations, coordination numbers, and support structures. We then summarize the microenvironment regulation in DACs, focusing on the elemental, electronic, geometric, and coordination effects between the two metal atom sites. These factors distinguish DACs from other catalysts and significantly enhance their catalytic reactivity and stability. The review also highlights recent contributions of DACs to HER, OER, and overall water electrolysis under different pH conditions, particularly emphasizing the microenvironment regulation of the dual-atom site effects on DACs for promoting water electrolysis. While there have been exciting developments in DACs, challenges remain in optimizing their syntheses, characterizations, and applications.

(1) Optimal selection of metal atom species and support materials is essential to catalyze specific reactions. This necessitates a comprehensive understanding of the complex interactions among metal atoms, as well as between atoms and supports. Such understanding is crucial in determining the catalytic performance, ensuring that these interactions mutually enhance reactivity. Besides, support materials with larger surface areas improve the catalytic selectivity and activity, while they may also possess irregular cavities or defects that potentially disrupt site equilibrium.

(2) Achieving a uniform distribution of diatomic sites remains a significant challenge. Many reported diatomic catalysts (DACs) fail to exhibit the anticipated uniform distribution of active sites, primarily due to their random spatial arrangement. This randomness can lead to aggregation or migration of active metal species, undermining their effectiveness. To address these issues, it is essential to develop and refine synthesis methods that consistently yield well-dispersed DACs.

(3) *In situ* characterization is necessary for understanding the properties of active sites, but it has certain limitations. The state of active sites is influenced by various factors such as adsorbates, current density, and high potentials. However, in most studies, powerful characterization techniques are typically employed in a non-*in situ* manner, such as XPS, HAADF-STEM, and XAFS. Although theoretical approaches like DFT calculations and simulations can provide theoretical insights, a significant disparity exists between theoretical understanding and practical application, making it challenging to fully comprehend the structure and composition of catalysts at the atomic scale. Therefore, suitable *in situ* characterization techniques for DACs are urgently needed.

(4) Sustaining the stability of DACs presents a significant challenge. The metal atoms within DACs are vulnerable to environmental conditions, including aggregation or deactivation of active sites in acidic environments and deactivation at high temperatures. These factors considerably impact both the catalytic activity and stability of DACs, thereby restricting their long-term applicability and recyclability.

In conclusion, this review aims to provide a comprehensive insight into DACs applied for water electrolysis. We



summarized the advancements in categorizing DACs, examined how regulating their microenvironment influences their properties, focused on the dual-atom site effect, and discussed their advanced applications in electrochemical water electrolysis. Despite facing challenges such as exploring the diversity of supports, precisely characterizing and modulating active structures, and fulfilling the requirements for large-scale practical applications, DACs present a promising avenue for exploring high-performance catalysts at the atomic level. Moreover, studying microenvironment regulation in DACs for water electrolysis can also provide references and guidance for other electrochemical applications.

## Data availability

Data availability does not apply to this article as no new data were created or analyzed in this study.

## Author contributions

X. R. and S. R. contributed equally to this work, which involved writing, editing, and finalizing the manuscript. G. J. conceived this work and co-wrote the manuscript. X. H. Q., X. G. and G. J. revised the manuscript. All authors read and approved the final manuscript.

## Conflicts of interest

There are no conflicts of interest to declare.

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## References

- 1 M. Chatenet, B. G. Pollet, D. R. Dekel, F. Dionigi, J. Deseure, P. Millet, R. D. Braatz, M. Z. Bazant, M. Eikerling, I. Staffell, P. Balcombe, Y. Shao-Horn and H. Schäfer, *Chem. Soc. Rev.*, 2022, **51**, 4583–4762.
- 2 K. Kohse-Hoeinghaus, *Chem. Rev.*, 2023, **123**, 5139–5219.
- 3 V. H. Do and J. M. Lee, *Chem. Soc. Rev.*, 2024, **53**, 2693–2737.
- 4 H. Nishiyama, T. Yamada, M. Nakabayashi, Y. Maehara, M. Yamaguchi, Y. Kuromiya, Y. Nagatsuma, H. Tokudome, S. Akiyama, T. Watanabe, R. Narushima, S. Okunaka, N. Shibata, T. Takata, T. Hisatomi and K. Domen, *Nature*, 2021, **598**, 304–307.
- 5 D. Zhao, Z. W. Zhuang, X. Cao, C. Zhang, Q. Peng, C. Chen and Y. D. Li, *Chem. Soc. Rev.*, 2020, **49**, 2215–2264.
- 6 Y. Wang, D. S. Wang and Y. D. Li, *Adv. Mater.*, 2021, **33**, 2008151.
- 7 W. Y. Zhang, Y. G. Chao, W. S. Zhang, J. H. Zhou, F. Lv, K. Wang, F. X. Lin, H. Luo, J. Li, M. P. Tong, E. K. Wang and S. J. Guo, *Adv. Mater.*, 2021, **33**, 2102576.
- 8 X. Sun, S. Wang, Y. Hou, X. F. Lu, J. Zhang and X. Wang, *J. Mater. Chem. A*, 2023, **11**, 13089–13106.
- 9 C. C. Hou, H. F. Wang, C. X. Li and Q. Xu, *Energy Environ. Sci.*, 2020, **13**, 1658–1693.
- 10 T. C. Pu, J. Q. Ding, F. X. Zhang, K. Wang, N. Cao, E. J. M. Hensen and P. F. Xie, *Angew. Chem., Int. Ed.*, 2023, **62**, e202305964.
- 11 W. T. Jing, H. Shen, R. X. Qin, Q. Y. Wu, K. L. Liu and N. F. Zheng, *Chem. Rev.*, 2023, **123**, 5948–6002.
- 12 Y. Chen, J. Lin, B. H. Jia, X. D. Wang, S. Y. Jiang and T. Y. Ma, *Adv. Mater.*, 2022, **34**, 2201796.
- 13 Y. X. Wang, X. Z. Cui, J. Q. Zhang, J. L. Qiao, H. T. Huang, J. L. Shi and G. X. Wang, *Prog. Mater. Sci.*, 2022, **128**, 100964.
- 14 W. Y. Zhang, S. Y. Yi, Y. H. Yu, H. Liu, A. Kucernak, J. Wu and S. Li, *J. Mater. Chem. A*, 2023, **12**, 87–112.
- 15 J. J. Ge, Z. J. Li, X. Hong and Y. D. Li, *Chem. - Eur. J.*, 2019, **25**, 5113–5127.
- 16 L. B. Li, K. Yuan and Y. W. Chen, *Acc. Mater. Res.*, 2022, **3**, 584–596.
- 17 Y. F. Hu, Z. S. Li, B. L. Li and C. L. Yu, *Small*, 2022, **18**, 2203589.
- 18 G. Rocha, M. A. R. da Silva, A. Rogolino, G. A. A. Diab, L. F. G. Noleto, M. Antonietti and I. F. Teixeira, *Chem. Soc. Rev.*, 2023, **52**, 4878–4932.
- 19 T. M. Tang, Z. L. Wang and J. Q. Guan, *Coord. Chem. Rev.*, 2023, **492**, 215288.
- 20 L. C. Liu and A. Corma, *Chem. Rev.*, 2023, **123**, 4855–4933.
- 21 A. Pedersen, J. Barrio, A. L. Li, R. Jervis, D. J. L. Brett, M. M. Titirici and I. E. L. Stephens, *Adv. Energy Mater.*, 2022, **12**, 2102715.
- 22 M. Balamurugan, H. Y. Jeong, V. S. K. Choutipalli, J. S. Hong, H. Seo, N. Saravanan, J. H. Jang, K. G. Lee, Y. H. Lee, S. W. Im, V. Subramanian, S. H. Kim and K. T. Nam, *Small*, 2020, **16**, 2000955.
- 23 J. Q. Shan, C. Ye, Y. L. Jiang, M. Jaroniec, Y. Zheng and S. Z. Qiao, *Sci. Adv.*, 2022, **8**, eabo0762.
- 24 Y. F. Zhao, H. Zhou, X. R. Zhu, Y. T. Qu, C. Xiong, Z. G. Xue, Q. W. Zhang, X. K. Liu, F. Y. Zhou, X. M. Mou, W. Y. Wang, M. Chen, Y. Xiong, X. G. Lin, Y. Lin, W. X. Chen, H. J. Wane, Z. Jiang, L. R. Zheng, T. Yao, J. C. Dong, S. Q. Wei, W. X. Huang, L. Gu, J. Luo, Y. F. Li and Y. E. Wu, *Nat. Catal.*, 2021, **4**, 134–143.
- 25 X. Hai, Y. Zheng, Q. Yu, N. Guo, S. B. Xi, X. X. Zhao, S. Mitchell, X. H. Luo, V. Tulus, M. Wang, X. Y. Sheng, L. B. Ren, X. D. Long, J. Li, P. He, H. H. Lin, Y. G. Cui, X. N. Peng, J. W. Shi, J. Wu, C. Zhang, R. Q. Zou, G. Guillén-Gosálbez, J. Pérez-Ramírez, M. J. Koh, Y. Zhu, J. Li and J. Lu, *Nature*, 2023, **622**, 754–760.
- 26 Y. Gao, B. Z. Liu and D. S. Wang, *Adv. Mater.*, 2023, **35**, 2209654.
- 27 J. X. Wang, D. N. Deng, Q. M. Wu, M. J. Liu, Y. C. Wang, J. B. Jiang, X. R. Zheng, H. R. Zheng, Y. Bai, Y. B. Chen, X. Xiong and Y. P. Lei, *ACS Nano*, 2023, **17**, 18688–18705.
- 28 X. N. Li, L. H. Liu, X. Y. Ren, J. J. Gao, Y. Q. Huang and B. Liu, *Sci. Adv.*, 2020, **6**, eabb6833.
- 29 Y. Chen, J. Lin, Q. Pan, X. Liu, T. Y. Ma and X. D. Wang, *Angew. Chem.*, 2023, **135**, e202306469.

- 30 Y. J. Sa and S. H. Joo, *Chem*, 2019, **5**, 3006–3007.
- 31 J. M. Wang, E. Kim, D. P. Kumar, A. P. Rangappa, Y. Kim, Y. X. Zhang and T. K. Kim, *Angew. Chem.*, 2022, **134**, e202113044.
- 32 P. Zhang, M. Gan, Y. Song, P. Liu, H. Liang, Y. Shen, B. Xu, T. Liao, J. Guo and Z. Sun, *Appl. Catal., B*, 2024, **355**, 124214.
- 33 Q. Hao, H. X. Zhong, J. Z. Wang, K. H. Liu, J. M. Yan, Z. H. Ren, N. Zhou, X. Zhao, H. Zhang, D. X. Liu, X. Liu, L. W. Chen, J. Luo and X. B. Zhang, *Nat. Synth.*, 2022, **1**, 719–728.
- 34 Y. S. Wei, L. M. Sun, M. Wang, J. H. Hong, L. L. Zou, H. W. Liu, Y. Wang, M. Zhang, Z. Liu, Y. W. Li, S. Horike, K. Suenaga and Q. Xu, *Angew. Chem., Int. Ed.*, 2020, **59**, 16013–16022.
- 35 N. Q. Zhang, X. X. Zhang, Y. K. Kang, C. L. Ye, R. Jin, H. Yan, R. Lin, J. R. Yang, Q. Xu, Y. Wang, Q. H. Zhang, L. Gu, L. C. Liu, W. Y. Song, J. Liu, D. S. Wang and Y. D. Li, *Angew. Chem.*, 2021, **133**, 13500–13505.
- 36 T. M. Tang, X. Bai, Z. L. Wang and J. Q. Guan, *Chem. Sci.*, 2024, **15**, 5082–5112.
- 37 W. J. Zou, R. H. Lu, X. L. Liu, G. F. Xiao, X. B. Liao, Z. Y. Wang and Y. Zhao, *J. Mater. Chem. A*, 2022, **10**, 9150–9160.
- 38 H. M. Chen, Y. Yang, C. Jiao, Z. W. Zhuo, J. J. Mao and Y. Liu, *Nano Res.*, 2023, **17**, 3413–3422.
- 39 F. F. Xu, B. Feng, Z. Shen, Y. Q. Chen, L. Jiao, Y. Zhang, J. Y. Tian, J. R. Zhang, X. Z. Wang, L. J. Yang, Q. Wu and Z. Hu, *J. Am. Chem. Soc.*, 2024, **146**, 9365–9374.
- 40 X. K. Wang, L. L. Xu, C. Li, C. H. Zhang, H. X. Yao, R. Xu, P. X. Cui, X. S. Zheng, M. Gu, J. Lee, H. Q. Jiang and M. H. Huang, *Nat. Commun.*, 2023, **14**, 7210.
- 41 H. W. Zhang, X. D. Jin, J. M. Lee and X. Wang, *ACS Nano*, 2022, **16**, 17572–17592.
- 42 Y. Z. Li, Y. J. Li, H. Sun, L. Y. Gao, X. R. Jin, Y. P. Li, Z. Lv, L. J. Xu, W. Liu and X. M. Sun, *Nano-Micro Lett.*, 2024, **16**, 139.
- 43 L. P. Wang, Z. H. Wei, Z. Y. Sun, L. L. Zhu, Y. Gao, Z. Chen, S. H. Li and W. X. Chen, *J. Mater. Chem. A*, 2024, **12**, 11749–11770.
- 44 J. Wang, Z. Q. Huang, W. Liu, C. R. Chang, H. L. Tang, Z. J. Li, W. X. Chen, C. J. Jia, T. Yao, S. Q. Wei, Y. Wu and Y. D. Lie, *J. Am. Chem. Soc.*, 2017, **139**, 17281–17284.
- 45 X. Sun, Y. Qiu, B. Jiang, Z. Y. Chen, C. H. Zhao, H. Zhou, L. Yang, L. S. Fan, Y. Zhang and N. Q. Zhang, *Nat. Commun.*, 2023, **14**, 291.
- 46 J. Chen, D. J. Zhang, B. Liu, K. Zheng, Y. F. Li, Y. B. Xu, Z. J. Li and X. H. Liu, *Angew. Chem., Int. Ed.*, 2024, **63**, e202401168.
- 47 Q. P. Zhao, W. X. Shi, J. W. Zhang, Z. Y. Tian, Z. M. Zhang, P. Zhang, Y. Wang, S. Z. Qiao and T. B. Lu, *Nat. Synth.*, 2024, **3**, 497–506.
- 48 H. S. Fan, X. Y. Liang, F. X. Ma, G. B. Zhang, Z. Q. Liu, L. Zhen, X. C. Zeng and C. Y. Xu, *Small*, 2024, **20**, 2307863.
- 49 F. H. K. Pan, T. Jin, W. W. Yang, H. Li, Y. Q. Cao, J. Hu, X. G. Zhou, H. L. Liu and X. Z. Duan, *Chem Catal.*, 2021, **1**, 734–745.
- 50 S. Zhang, J. H. Wu, M. T. Zheng, X. Jin, Z. H. Shen, Z. H. Li, Y. J. Wang, Q. Wang, X. B. Wang, H. Wei, J. W. Zhang, P. Wang, S. Q. Zhang, L. Y. Yu, L. F. Dong, Q. S. Zhu, H. G. Zhang and J. Lu, *Nat. Commun.*, 2023, **14**, 3634.
- 51 Z. J. Li, S. Q. Ji, C. Xu, L. P. Leng, H. X. Liu, J. H. Horton, L. Du, J. C. Gao, C. He, X. Y. Qi, Q. Xu and J. F. Zhu, *Adv. Mater.*, 2023, **35**, 2209644.
- 52 R. Z. Li, Z. D. Zhang, X. Liang, J. Shen, J. Wang, W. M. Sun, D. S. Wang, J. C. Jiang and Y. D. Li, *J. Am. Chem. Soc.*, 2023, **145**, 16218–16227.
- 53 X. Zhao, W. P. Li, Y. H. Cao, A. Portniagin, B. Tang, S. X. Wang, Q. Liu, D. Y. W. Yu, X. Y. Zhong, X. R. Zheng and A. L. Rogach, *ACS Nano*, 2024, **18**, 4256–4268.
- 54 Z. P. Yu, C. W. Si, A. P. LaGrow, Z. X. Tai, W. A. Caliebe, A. Tayal, M. J. Sampaio, J. P. S. Sousa, I. Amorim, A. Araujo, L. J. Meng, J. L. Faria, J. Y. Xu, B. Li and L. F. Liu, *ACS Catal.*, 2022, **12**, 9397–9409.
- 55 W. C. Wan, Y. G. Zhao, S. Q. Wei, C. A. Triana, J. G. Li, A. Arcifa, C. S. Allen, R. Cao and G. R. Patzke, *Nat. Commun.*, 2021, **12**, 5589.
- 56 Y. X. Zhang, S. B. Zhang, H. L. Huang, X. L. Liu, B. B. Li, Y. Lee, X. D. Wang, Y. Bai, M. Z. Sun, Y. F. Wu, S. Y. Gong, X. W. Liu, Z. B. Zhuang, T. Tan and Z. Q. Niu, *J. Am. Chem. Soc.*, 2023, **145**, 4819–4827.
- 57 T. T. Cui, Y. P. Wang, T. Ye, J. Wu, Z. Q. Chen, J. Li, Y. P. Lei, D. S. Wang and Y. D. Li, *Angew. Chem., Int. Ed.*, 2022, **61**, e202115219.
- 58 J. D. Yi, X. P. Gao, H. Zhou, W. Chen and Y. E. Wu, *Angew. Chem., Int. Ed.*, 2022, **61**, e202212329.
- 59 H. N. Shi, H. Z. Wang, Y. C. Zhou, J. H. Li, P. L. Zhai, X. Y. Li, G. G. Gurzadyan, J. G. Hou, H. Yang and X. W. Guo, *Angew. Chem.*, 2022, **134**, e202208904.
- 60 Y. C. Chai, S. H. Chen, Y. Chen, F. F. Wei, L. R. Cao, J. Lin, L. Li, X. Y. Liu, S. Lin, X. D. Wang and T. Zhang, *J. Am. Chem. Soc.*, 2023, **146**, 263–273.
- 61 D. D. Wang, X. R. Zhu, X. J. Tu, X. R. Zhang, C. Chen, X. X. Wei, Y. F. Li and S. Y. Wang, *Adv. Mater.*, 2023, **35**, 2304646.
- 62 B. W. Wang, J. X. Zou, X. C. Shen, Y. C. Yang, G. Z. Hu, W. Li, Z. M. Peng, D. Banham, A. G. Dong and D. Y. Zhao, *Nano Energy*, 2019, **63**, 103851.
- 63 Z. Wang, R. J. Xu, Q. T. Ye, X. Y. Jin, Z. Lu, Z. B. Yang, Y. Wang, T. Yan, Y. P. Liu, Z. J. Pan, S. J. Hwang and H. J. Fan, *Adv. Funct. Mater.*, 2024, 2315376, DOI: [10.1002/adfm.202315376](https://doi.org/10.1002/adfm.202315376).
- 64 J. Y. Chen, H. Li, C. Fan, Q. W. Meng, Y. W. Tang, X. Y. Qiu, G. T. Fu and T. Y. Ma, *Adv. Mater.*, 2020, **32**, 2003134.
- 65 L. X. Wang, X. P. Gao, S. C. Wang, C. Chen, J. Song, X. H. Ma, T. Yao, H. Zhou and Y. Wu, *J. Am. Chem. Soc.*, 2023, **145**, 13462–13468.
- 66 Z. J. Li, S. Q. Ji, C. Wang, H. X. Liu, L. P. Leng, L. Du, J. C. Gao, M. Qiao, J. H. Horton and Y. Wang, *Adv. Mater.*, 2023, **35**, 2300905.
- 67 J. Q. Jiao, Q. Yuan, M. J. Tan, X. Q. Han, M. B. Gao, C. Zhang, X. Yang, Z. L. Shi, Y. B. Ma, H. Xiao, J. W. Zhang and T. B. Lu, *Nat. Commun.*, 2023, **14**, 6164.



- 68 Y. Yang, B. Li, Y. N. Liang, W. P. Ni, X. Li, G. Z. Shen, L. Xu, Z. J. Chen, C. Zhu, J. X. Liang and S. G. Zhang, *Adv. Sci.*, 2024, **11**, 2310231.
- 69 A. Kumar, S. Ibraheem, T. A. Nguyen, R. K. Gupta, T. Maiyalagan and G. Yasin, *Coord. Chem. Rev.*, 2021, **446**, 214122.
- 70 D. Grammatico, A. J. Bagnall, L. Riccardi, M. Fontecave, B. L. Su and L. Billon, *Angew. Chem.*, 2022, **134**, e202206399.
- 71 R. Jasinski, *Nature*, 1964, **201**, 1212–1213.
- 72 W. B. Liu, K. Wang, C. M. Wang, W. P. Liu, H. H. Pan, Y. J. Xiang, D. D. Qi and J. Z. Jiang, *J. Mater. Chem. A*, 2018, **6**, 22851–22857.
- 73 S. Yuan, J. Y. Peng, Y. R. Zhang, D. J. Zheng, S. Bagi, T. Wang, Y. Román-Leshkov and Y. Shao-Horn, *ACS Catal.*, 2022, **12**, 7278–7287.
- 74 M. R. Narouz, P. De La Torre, L. An and C. J. Chang, *Angew. Chem., Int. Ed.*, 2022, **61**, e202207666.
- 75 Y. F. Zhang, Z. X. Yu, F. X. She, L. Wei, Z. Y. Zeng and H. Li, *J. Colloid Interface Sci.*, 2023, **640**, 983–989.
- 76 Z. Zhang, X. J. Ma, Y. M. Li, N. N. Ma, M. Wang, W. Liu, J. H. Peng, Y. W. Liu and Y. D. Li, *J. Am. Chem. Soc.*, 2024, **146**, 8425–8434.
- 77 W. X. Zhang, M. R. Zhang, H. J. Wang, W. Zhang and M. Zhang, *Nano Res.*, 2024, **17**, 4850–4855.
- 78 L. Bai, C.-S. Hsu, D. T. Alexander, H. M. Chen and X. Hu, *Nat. Energy*, 2021, **6**, 1054–1066.
- 79 J. Ballesteros-Soberanas, N. Martín, M. Bacic, E. Tiburcio, M. Mon, J. C. Hernández-Garrido, C. Marini, M. Boronat, J. Ferrando-Soria, D. Armentano, E. Pardo and A. Leyva-Pérez, *Nat. Catal.*, 2024, **7**, 452–463.
- 80 S. Samireddi, V. Aishwarya, I. Shown, S. Muthusamy, S. M. Unni, K. T. Wong, K. H. Chen and L. C. Chen, *Small*, 2021, **17**, 2103823.
- 81 A. Kumar, K. Sun, X. X. Duan, S. B. Tian and X. M. Sun, *Chem. Mater.*, 2022, **34**, 5598–5606.
- 82 Z. Y. Lu, B. F. Wang, Y. F. Hu, W. Liu, Y. F. Zhao, R. O. Yang, Z. P. Li, J. Luo, B. Chi, Z. Jiang, M. S. Li, S. C. Mu, S. J. Liao, J. J. Zhang and X. L. Sun, *Angew. Chem.*, 2019, **131**, 2648–2652.
- 83 J. Wang, C. X. Zhao, J. N. Liu, Y. W. Song, J. Q. Huang and B. Q. Li, *Nano Energy*, 2022, **104**, 107927.
- 84 J. C. Hao, Z. C. Zhuang, J. C. Hao, C. Wang, S. L. Lu, F. Duan, F. P. Xu, M. L. Du and H. Zhu, *Adv. Energy Mater.*, 2022, **12**, 2200579.
- 85 H. M. Liu, Q. L. Huang, W. An, Y. Q. Wang, Y. Men and S. Liu, *J. Energy Chem.*, 2021, **61**, 507–516.
- 86 M. Y. Jiao, Z. P. Chen, N. L. Wang and L. C. Liu, *Appl. Catal., B*, 2023, **324**, 122244.
- 87 X. M. Zhang, Y. Li, M. Jiang, J. X. Wei, X. X. Ding, C. Y. Zhu, H. He, H. C. Lai and J. Y. Shi, *Chem. Eng. J.*, 2021, **426**, 130758.
- 88 M. M. Tong, F. F. Sun, Y. Xie, Y. Wang, Y. Q. Yang, C. G. Tian, L. Wang and H. G. Fu, *Angew. Chem., Int. Ed.*, 2021, **60**, 14005–14012.
- 89 C. Fang, J. Zhou, L. L. Zhang, W. C. Wan, Y. X. Ding and X. Y. Sun, *Nat. Commun.*, 2023, **14**, 4449.
- 90 X. Liu, Y. Jiao, Y. Zheng, K. Davey and S. Z. Qiao, *J. Mater. Chem. A*, 2019, **7**, 3648–3654.
- 91 A. M. Roth-Zawadzki, A. J. Nielsen, R. E. Tankard and J. Kibsgaard, *ACS Catal.*, 2024, **14**, 1121–1145.
- 92 K. Wang, L. Y. Wang, J. R. Huang, Y. Chen, X. P. Liu, T. F. Yang, G. Y. Wei and S. Y. Gao, *Nanoscale*, 2023, **15**, 18395–18406.
- 93 H. M. Liu, H. P. Rong and J. T. Zhang, *Chemsuschem*, 2022, **15**, e202200498.
- 94 Y. Y. Zhao, K. R. Yang, Z. C. Wang, X. X. Yan, S. F. Cao, Y. F. Ye, Q. Dong, X. Z. Zhang, J. E. Thorne, L. Jin, K. L. Materna, A. Trimpalis, H. Y. Bai, S. C. Fakra, X. Y. Zhong, P. Wang, X. Q. Pan, J. H. Guo, M. Flytzani-Stephanopoulos, G. W. Brudvig, V. S. Batista and D. W. Wang, *Proc. Natl. Acad. Sci. U. S. A.*, 2018, **115**, 2902–2907.
- 95 B. T. Hu, A. J. Huang, X. J. Zhang, Z. Chen, R. Y. Tu, W. Zhu, Z. B. Zhuang, C. Chen, Q. Peng and Y. D. Li, *Nano Res.*, 2021, **14**, 3482–3488.
- 96 W. Wang, L. Shang, G. J. Chang, C. Y. Yan, R. Shi, Y. X. Zhao, G. I. N. Waterhouse, D. J. Yang and T. R. Zhang, *Adv. Mater.*, 2019, **31**, 1808276.
- 97 Z. Y. Zhang, X. X. Zhao, S. B. Xi, L. L. Zhang, Z. X. Chen, Z. P. Zeng, M. Huang, H. B. Yang, B. Liu, S. J. Pennycook and P. Chen, *Adv. Energy Mater.*, 2020, **10**, 2002896.
- 98 R. J. Gao, J. S. Xu, J. Wang, J. Lim, C. Peng, L. Pan, X. W. Zhang, H. M. Yang and J. J. Zou, *J. Am. Chem. Soc.*, 2022, **144**, 573–581.
- 99 C. L. Chen, M. Z. Sun, F. Zhang, H. J. Li, M. R. Sun, P. Fang, T. L. Song, W. X. Chen, J. C. Dong, B. Rosen, P. W. Chen, B. L. Huang and Y. J. Li, *Energy Environ. Sci.*, 2023, **16**, 1685–1696.
- 100 Q. Wang, B. J. Jin, M. Hu, C. Y. Jia, X. Li, E. Sharman and J. Jiang, *J. Phys. Chem. C*, 2021, **125**, 5616–5622.
- 101 T. Zhang, Y. Liu, L. Xue, J. Sun, P. Xiong and J. Zhu, *J. Energy Chem.*, 2024, **94**, 273–287.
- 102 G. F. Hu, L. Shang, T. Sheng, Y. G. Chen and L. Y. Wang, *Adv. Funct. Mater.*, 2020, **30**, 2002281.
- 103 D. Z. Yao, C. Tang, X. Zhi, B. Johannessen, A. Slattery, S. Chern and S. Z. Qiao, *Adv. Mater.*, 2023, **35**, 2209386.
- 104 Z. Y. Jin, P. P. Li, Y. Meng, Z. W. Fang, D. Xiao and G. H. Yu, *Nat. Catal.*, 2021, **4**, 615–622.
- 105 S. Zhang, J. Huang, L. Ma, D. Zhai, B. Wei, H. Yang and C. He, *Energy Rev.*, 2024, **3**, 100075.
- 106 R. Cepitis, N. Kongi, J. Rossmeisl and V. Ivaništšev, *ACS Energy Lett.*, 2023, **8**, 1330–1335.
- 107 R. Cepitis, N. Kongi, J. Rossmeisl and V. Ivanistsev, *ACS Energy Lett.*, 2023, **8**, 1330–1335.
- 108 F. Calle-Vallejo, J. I. Martínez and J. Rossmeisl, *Phys. Chem. Chem. Phys.*, 2011, **13**, 15639–15643.
- 109 Y. N. Meng, C. Yin, K. Li, H. Tang, Y. Wang and Z. J. Wu, *ACS Sustainable Chem. Eng.*, 2019, **7**, 17273–17281.
- 110 H. Wan, T. M. Ostergaard, L. Arnarson and J. Rossmeisl, *ACS Sustainable Chem. Eng.*, 2019, **7**, 611–617.
- 111 Y. Yan, H. Y. Cheng, Z. H. Qu, R. Yu, F. Liu, Q. W. Ma, S. Zhao, H. Hu, Y. Cheng, C. Y. Yang, Z. F. Li, X. Wang,

- S. Y. Hao, Y. Y. Chen and M. K. Liu, *J. Mater. Chem. A*, 2021, **9**, 19489–19507.
- 112 J. R. Yang, W. H. Li, D. S. Wang and Y. D. Li, *Adv. Mater.*, 2020, **32**, 2003300.
- 113 R. M. Hu, Y. C. Li, Q. W. Zeng and J. X. Shang, *Appl. Surf. Sci.*, 2020, **525**, 146588.
- 114 D. H. Deng, X. Q. Chen, L. Yu, X. Wu, Q. F. Liu, Y. Liu, H. X. Yang, H. F. Tian, Y. F. Hu, P. P. Du, R. Si, J. H. Wang, X. J. Cui, H. B. Li, J. P. Xiao, T. Xu, J. Deng, F. Yang, P. N. Duchesne, P. Zhang, J. G. Zhou, L. T. Sun, J. Q. Li, X. L. Pan and X. H. Bao, *Sci. Adv.*, 2015, **1**, e1500462.
- 115 H. L. Huang, M. Z. Sun, S. W. Li, S. B. Zhang, Y. Y. Lee, Z. W. Li, J. J. Fang, C. J. Chen, Y. X. Zhang, Y. F. Wu, Y. Z. Che, S. R. Qian, W. Zhu, C. Tang, Z. B. Zhuang, L. Zhang and Z. Q. Niu, *J. Am. Chem. Soc.*, 2024, **146**, 9434–9443.
- 116 Y. Wang, J. Wu, S. H. Tang, J. R. Yang, C. L. Ye, J. Chen, Y. P. Lei and D. S. Wang, *Angew. Chem., Int. Ed.*, 2023, **62**, e202219191.
- 117 B. Tang, Y. Zhou, Q. Ji, Z. Zhuang, L. Zhang, C. Wang, H. Hu, H. Wang, B. Mei and F. Song, *Nat. Synth.*, 2024, 1–13.
- 118 M. T. Jin, X. Zhang, S. Z. Niu, Q. Wang, R. Q. Huang, R. H. Ling, J. Q. Huang, R. Shi, A. Amini and C. Cheng, *ACS Nano*, 2022, **16**, 11577–11597.
- 119 C. L. Chen, M. R. Sun, K. X. Wang and Y. J. Li, *Smartmat*, 2022, **3**, 533–564.
- 120 J. C. Wilson, S. Caratzoulas, D. G. Vlachos and Y. Yan, *Nat. Commun.*, 2023, **14**, 2384.
- 121 J. X. Zhang, L. H. Zhang, J. M. Liu, C. Z. Zhong, Y. H. Tu, P. Li, L. Du, S. L. Chen and Z. M. Cui, *Nat. Commun.*, 2022, **13**, 5497.
- 122 T. T. Chao, X. Luo, W. X. Chen, B. Jiang, J. J. Ge, Y. Lin, G. Wu, X. Q. Wang, Y. M. Hu, Z. B. Zhuang, Y. E. Wu, X. Hong and Y. D. Li, *Angew. Chem., Int. Ed.*, 2017, **56**, 16047–16051.
- 123 L. Zhang, R. T. Si, H. S. Liu, N. Chen, Q. Wang, K. Adair, Z. Q. Wang, J. T. Chen, Z. X. Song, J. J. Li, M. N. Banis, R. Y. Li, T. K. Sham, M. Gu, L. M. Liu, G. A. Botton and X. L. Sun, *Nat. Commun.*, 2019, **10**, 4936.
- 124 Z. G. Chen, Y. F. Xu, D. Ding, G. Song, X. X. Gan, H. Li, W. Wei, J. Chen, Z. Y. Li, Z. M. Gong, X. M. Dong, C. F. Zhu, N. N. Yang, J. Y. Ma, R. Gao, D. Luo, S. Cong, L. Wang, Z. G. Zhao and Y. Cui, *Nat. Commun.*, 2022, **13**, 763.
- 125 C. Cai, K. Liu, L. Zhang, F. B. Li, Y. Tan, P. C. Li, Y. Q. Wang, M. Y. Wang, Z. X. Feng, D. M. Meira, W. Q. Qu, A. Stefancu, W. Z. Li, H. M. Li, J. W. Fu, H. Wang, D. S. Zhang, E. Cortés and M. Liu, *Angew. Chem., Int. Ed.*, 2023, **62**, e202300873.
- 126 W. K. Zhao, C. Luo, Y. Lin, G. B. Wang, H. M. Chen, P. Y. Kuang and J. G. Yu, *ACS Catal.*, 2022, **12**, 5540–5548.
- 127 Y. Zhou, E. H. Song, W. Chen, C. U. Segre, J. D. Zhou, Y. C. Lin, C. Zhu, R. G. Ma, P. Liu, S. F. Chu, T. Thomas, M. H. Yang, Q. Liu, K. Suenaga, Z. Liu, J. J. Liu and J. C. Wang, *Adv. Mater.*, 2020, **32**, 2003484.
- 128 R. B. Sun, W. X. Guo, X. Han and X. Hong, *Chem. Res. Chin. Univ.*, 2020, **36**, 597–610.
- 129 L. Zhang, K. Doyle-Davis and X. L. Sun, *Energy Environ. Sci.*, 2019, **12**, 492–517.
- 130 F. Y. Yu, Z. L. Lang, L. Y. Yin, K. Feng, Y. J. Xia, H. Q. Tan, H. T. Zhu, J. Zhong, Z. H. Kang and Y. G. Li, *Nat. Commun.*, 2020, **11**, 490.
- 131 J. Q. Zhang, Y. F. Zhao, X. Guo, C. Chen, C. L. Dong, R. S. Liu, C. P. Han, Y. D. Li, Y. Gogotsi and G. X. Wang, *Nat. Catal.*, 2018, **1**, 985–992.
- 132 M. J. Wang, Y. Xu, C. K. Peng, S. Y. Chen, Y. G. Lin, Z. W. Hu, L. Sun, S. Y. Ding, C. W. Pao, Q. Shao and X. Q. Huang, *J. Am. Chem. Soc.*, 2021, **143**, 16512–16518.
- 133 R. Yao, K. A. Sun, K. Y. Zhang, Y. Wu, Y. J. Du, Q. Zhao, G. Liu, C. Chen, Y. H. Sun and J. P. Li, *Nat. Commun.*, 2024, **15**, 2218.
- 134 Y. M. Da, Z. L. Tian, R. Jiang, Y. Liu, X. Lian, S. B. Xi, Y. Shi, Y. P. Wang, H. T. Lu, B. H. Cui, J. F. Zhang, X. P. Han, W. Chen and W. B. Hu, *Sci. China Mater.*, 2023, **66**, 1389–1397.
- 135 Y. Yang, Y. M. Qian, H. J. Li, Z. H. Zhang, Y. W. Mu, D. Do, B. Zhou, J. Dong, W. J. Yan, Y. Qin, L. Fang, R. F. Feng, J. G. Zhou, P. Zhang, J. C. Dong, G. H. Yu, Y. Y. Liu, X. M. Zhang and X. J. Fan, *Sci. Adv.*, 2020, **6**, eaba6586.
- 136 W. H. Fang, Y. Wu, S. X. Xin, Y. Hu, J. Q. Dang, M. Li, B. Z. Chen, H. Zhao and Z. X. Li, *Chem. Eng. J.*, 2023, **468**, 143605.
- 137 J. B. Shi, R. Li, J. T. Zhang, Y. Wang, W. L. Ma, Z. Y. Yue, C. H. Jin, Y. J. Liu, L. R. Zheng, J. B. Bai, X. L. Li, K. Y. Leng and Y. T. Qu, *ACS Appl. Mater. Interfaces*, 2023, **16**, 889–897.
- 138 H. N. Sun, X. M. Xu, Y. F. Song, W. Zhou and Z. P. Shao, *Adv. Funct. Mater.*, 2021, **31**, 2009779.
- 139 H. Y. Zhong, Q. Zhang, J. C. Yu, X. Zhang, C. Wu, Y. F. Ma, H. An, H. Wang, J. Zhang, X. P. Wang and J. M. Xue, *Adv. Energy Mater.*, 2023, **13**, 2301391.
- 140 Z. Pei, H. Zhang, Z.-P. Wu, X. F. Lu, D. Luan and X. W. Lou, *Sci. Adv.*, 2023, **9**, eadh1320.
- 141 Q. L. Ma and S. C. Mu, *Interdiscip. Mater.*, 2023, **2**, 53–90.
- 142 G. Gao, Z. Sun, X. Chen, G. Zhu, B. Sun, S. Liu and Y. Yamauchi, *Appl. Catal., B*, 2023, **343**, 123584.
- 143 C. L. Rong, K. Dastafkan, Y. Wang and C. Zhao, *Adv. Mater.*, 2023, **35**, 2211884.
- 144 T. Y. Zhang, J. J. Jiang, W. M. Sun, S. Y. Gong, X. W. Liu, Y. Tian and D. S. Wang, *Proc. Natl. Acad. Sci. U. S. A.*, 2024, **121**, e2317247121.
- 145 Q. L. Wu, K. Jiang, J. H. Han, D. C. Chen, M. Luo, J. Lan, M. Peng and Y. W. Tan, *Sci. China Mater.*, 2022, **65**, 1262–1268.
- 146 C. Lin, J. L. Li, X. P. Li, S. Yang, W. Luo, Y. J. Zhang, S. H. Kim, D. H. Kim, S. S. Shinde, Y. F. Li, Z. P. Liu, Z. Jiang and J. H. Lee, *Nat. Catal.*, 2021, **4**, 1012–1023.
- 147 D. F. Zhang, M. N. Li, X. Yong, H. Q. Song, G. I. N. Waterhouse, Y. F. Yi, B. J. Xue, D. L. Zhang, B. Z. Liu and S. Y. Lu, *Nat. Commun.*, 2023, **14**, 2517.
- 148 J. Liang, X. T. Gao, B. Guo, Y. Ding, J. W. Yan, Z. X. Guo, E. C. M. Tse and J. X. Liu, *Angew. Chem., Int. Ed.*, 2021, **60**, 12770–12774.



- 149 L. Han, S. Dong and E. Wang, *Adv. Mater.*, 2016, **28**, 9266–9291.
- 150 M. Yu, E. Budiyo and H. Tüysüz, *Angew. Chem., Int. Ed.*, 2022, **61**, e202103824.
- 151 S. Parvin, D. K. Chaudhary, A. Ghosh and S. Bhattacharyya, *ACS Appl. Mater. Interfaces*, 2019, **11**, 30682–30693.
- 152 M. Liu, N. Li, S. F. Cao, X. M. Wang, X. Q. Lu, L. J. Kong, Y. H. Xu and X. H. Bu, *Adv. Mater.*, 2022, **34**, 2107421.
- 153 Y. Zhou, L. Chen, L. Sheng, Q. Luo, W. Zhang and J. Yang, *Nano Res.*, 2022, **15**, 7994–8000.
- 154 Y. T. He, X. X. Yang, Y. S. Li, L. T. Liu, S. W. Guo, C. Y. Shu, F. Liu, Y. N. Liu, Q. Tan and G. Wu, *ACS Catal.*, 2022, **12**, 1216–1227.
- 155 Z. H. Pei, X. F. Lu, H. B. Zhang, Y. X. Li, D. Y. Luan and X. W. Lou, *Angew. Chem.*, 2022, **134**, e202207537.
- 156 L. She, G. Zhao, T. Ma, J. Chen, W. Sun and H. Pan, *Adv. Funct. Mater.*, 2022, **32**, 2108465.
- 157 X. Lin, J. Xu and Z. Peng, *Next Sustainability*, 2024, **3**, 100023.
- 158 L. Castanheira, W. O. Silva, F. H. Lima, A. Crisci, L. Dubau and F. Maillard, *ACS Catal.*, 2015, **5**, 2184–2194.
- 159 F. Zoller, S. Häring, D. Boehm, J. Luxa, Z. Sofer and D. Fattakhova-Rohlfing, *Small*, 2021, **17**, 2007484.
- 160 Y. C. Pi, Q. Shao, P. T. Wang, J. Guo and X. Q. Huang, *Adv. Funct. Mater.*, 2017, **27**, 1700886.
- 161 Z. Z. Jiang, W. D. Zhou, C. Hu, X. F. Luo, W. Zeng, X. G. Gong, Y. Yang, T. Yu, W. Lei and C. L. Yuan, *Adv. Mater.*, 2023, **35**, 2300505.
- 162 L. G. Wang, X. X. Duan, X. J. Liu, J. Gu, R. Si, Y. Qiu, Y. M. Qiu, D. E. Shi, F. H. Chen, X. M. Sun, J. H. Lin and J. L. Sun, *Adv. Energy Mater.*, 2020, **10**, 1903137.
- 163 D. S. Yu, Y. C. Ma, F. Hu, C. C. Lin, L. L. Li, H. Y. Chen, X. P. Han and S. J. Peng, *Adv. Energy Mater.*, 2021, **11**, 2101242.
- 164 H. C. Sun, C. W. Tung, Y. Qiu, W. Zhang, Q. Wang, Z. S. Li, J. Tang, H. C. Chen, C. D. Wang and H. M. Chen, *J. Am. Chem. Soc.*, 2022, **144**, 1174–1186.
- 165 D. Lebedev, M. Povia, K. Waltar, P. M. Abdala, I. E. Castelli, E. Fabbri, M. V. Blanco, A. Fedorov, C. Copéret, N. Marzari and T. J. Schmidt, *Chem. Mater.*, 2017, **29**, 5182–5191.
- 166 J. Wang, *Chem*, 2023, **9**, 1645–1657.
- 167 K. Yuan, X. D. Zhuang, H. Y. Fu, G. Brunklaus, M. Forster, Y. W. Chen, X. L. Feng and U. Scherf, *Angew Chem. Int. Ed. Engl.*, 2016, **55**, 6858–6863.
- 168 L. Wang, Y. H. Zhu, J. Q. Wang, F. D. Liu, J. F. Huang, X. J. Meng, J. M. Basset, Y. Han and F. S. Xiao, *Nat. Commun.*, 2015, **6**, 6957.
- 169 C. Rong, X. Shen, Y. Wang, L. Thomsen, T. Zhao, Y. Li, X. Lu, R. Amal and C. Zhao, *Adv. Mater.*, 2022, **34**, 2110103.