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## Corrosion inhibition performance of benzimidazole derivatives for protection of carbon steel in hydrochloric acid solution†

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This paper presents a comprehensive study on the corrosion inhibition properties of new organic compounds, (*1H*-benzimidazol-2-yl)methanethiol ( $LF_1$ ) and 1-dodecyl-2-((dodecylthio)methyl)-*1H*-benzimidazole ( $LF_2$ ), have been examined for inhibiting of Carbon-Steel (C.S) in 1.0 M HCl. Numerous methods, such as potentiodynamic polarization, electrochemical impedance spectroscopy, scanning electron microscopy (SEM) and Energy Dispersive X-ray (EDX) analysis, atomic force microscopy (AFM), contact angle measurements, UV-visible spectroscopy, and theoretical calculations, were used to evaluate the effectiveness in preventing corrosion. The two benzimidazoles ( $LF_1$  and  $LF_2$ )' inhibitory efficacy rose as their concentration increased, peaking at 88.2% and 95.4% respectively. The polarization graphs show a mixed behavior, with anodic predominance for  $LF_1$  and cathodic predominance for  $LF_2$ . Thermodynamic investigations showed that the values of  $\Delta G_{ads}$  were  $-40.0\text{ kJ mol}^{-1}$  for  $LF_1$  and  $-43.1\text{ kJ mol}^{-1}$  for  $LF_2$ , highlighting their strong adsorption onto the metal surface. Their adsorption process was in line with the Langmuir isotherm. Density Functional Theory (DFT) and Molecular Dynamics (MD) modeling have been utilized to examine and clarify the relationship between the inhibitor and carbon steel (C.S).

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## 1. Introduction

Carbon steel is often utilized as a construction material in various industrial applications due to its cost-effectiveness, favorable physicochemical properties, and widespread availability. As indicated by the references, this encompasses tasks such as constructing pipelines for the transportation of gas and oil, acid descaling, and acid pickling.<sup>1,2</sup>

On the other hand, if the same material has to be cleaned and descaled using a strong acidic solution, especially 1.0 M HCl, it can be readily damaged, resulting in economic and human calamities. Because of this, this phenomenon is a critical issue that has scholars' attention. In order to reduce the corrosive assault on metallic surfaces, corrosion inhibitors, which are frequently used at low concentrations in acidic situations, are essential for protecting metals. Organic compounds having polar functionalities comprising nitrogen, sulfur, and/or oxygen in the conjugated system have been found to have

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effective inhibitory capabilities among alternative corrosion inhibitors.<sup>3,4</sup> Research indicates that compounds containing N, S, and O-azoles, pyrrole, quinoline, and their derivatives, for example-have significant anti-corrosion properties.<sup>5,6</sup>

The characteristics of the corrosive media, the inhibitor's chemical structure, and the metal surface all affect adsorption.<sup>7</sup> The interaction between the investigated metal and organic molecules has been characterized using two different mechanisms: chemical adsorption and physisorption. The kind and concentration of the medium, along with its temperature, are the main factors that influence the choice of appropriate inhibitors. Numerous writers have investigated how benzimidazole and its derivatives-especially N-heterocyclic compounds-affect carbon steel corrosion in acidic environments.<sup>8</sup> Numerous researchers frequently employ benzimidazole bases as corrosion inhibitors. The benzimidazole derivatives demonstrated outstanding steel inhibition effectiveness in acidic solution, according to recent investigations.<sup>9-13</sup>

Benzimidazole and its many derivatives have attracted a great deal of interest in the field of metal corrosion protection. Several studies have demonstrated that these compounds are excellent inhibitors, particularly in acidic environments.<sup>14-17</sup> Previous research has examined benzimidazole and its derivatives as corrosion inhibitors for carbon steel in a 1 M hydrochloric acid solution. These studies revealed that all three benzimidazole derivatives exhibited satisfactory corrosion inhibition efficacy.<sup>14</sup> One study combined experimental and quantum chemical approaches to evaluate the inhibition performance of certain benzimidazole derivatives on mild steel corrosion in HCl, highlighting their effectiveness as corrosion inhibitors.

Benzimidazole and its multiple derivatives have also been studied in biochemical and pharmacological contexts.<sup>18</sup> Literature searches have confirmed that benzimidazole molecules are considered a promising class of bioactive heterocyclic compounds, renowned for their efficacy against various strains of microorganisms. Benzimidazole derivatives exhibit a wide range of activities, including antimicrobial,<sup>19</sup> antiparasitic,<sup>20</sup> anti-HIV,<sup>21</sup> and anticancer.<sup>22</sup>

The current study assesses the inhibitory efficacy of two synthetic benzimidazole compounds, (1*H*-benzimidazol-2-yl)methanethiol (**LF<sub>1</sub>**) and 1-dodecyl-2-[(dodecylsulfanyl)methyl]-1*H*-1,3-benzimidazole (**LF<sub>2</sub>**), against corrosion of C.S in 1.0 M

HCl solution. The inhibition efficiency of these organic compounds was successively evaluated using potentiodynamic polarization curves, electrochemical impedance spectroscopy, isotherm calculations, SEM-EDX, AFM, contact angle, and UV-visible analysis, the efficiency of these organic compounds' inhibition was successively evaluated. DFT computations and molecular dynamics simulations (MDS) were used to determine the relationship between the investigated derivatives' molecular structures and their corrosion inhibition characteristics. Also proposed and covered was the adsorption mechanism. Thermodynamic and adsorption characteristics were also computed. Table 1 displays the molecular structures of the **LF<sub>1</sub>** and **LF<sub>2</sub>** under investigation.

## 2. Experimental

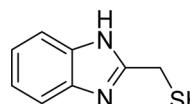
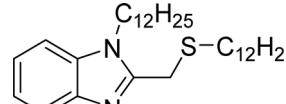
### 2.1. Materials

The chemical composition (in weight percentage) of the carbon steel (C.S) specimens used in this investigation is as follows: C-0.3700%, Si-0.2300%, Mn-0.6800%, S-0.0160%, Cr-0.0770%, Ti-0.0110%, Ni-0.0590%, Co-0.0090%, Cu-0.1600%, and iron (Fe) as the remaining content. SiC emery grade (180–1200) papers were used to polish these samples. After that, they were cleaned with distilled water, degreased with acetone, and dried with hot air. Distilled water was used to dilute concentrated acid (37.0%), which had a density of  $d = 1.18$ , to create the aggressive solution.

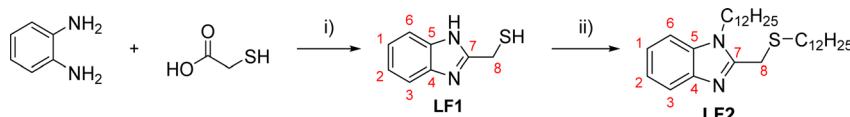
### 2.2. Synthesis of **LF<sub>1</sub>** and **LF<sub>2</sub>**

The general method for the synthesis of 2-substituted benzimidazoles was based on the Phillips procedure.<sup>23</sup> The condensation of commercially available *o*-phenylenediamine (0.02 mol) and thioglycolic acid (0.03 mol), was conducted in the presence of boiling 4 N hydrochloric acid (20 mL) under reflux for 2 h, as shown in Scheme 1. The reaction mixture was cooled and neutralized using a sodium hydroxide solution. The crude product was washed with cold distilled water, dissolved in boiling water for recrystallization, and finally filtered and dried to obtain the desired **LF<sub>1</sub>** product with a darkish aspect. Yield: 60%; <sup>1</sup>H-NMR (600 MHz, DMSO-d<sub>6</sub>)  $\delta$  (ppm), 7.56–7.18 (m, 4H), 4.22 (s, 2H). <sup>13</sup>C-NMR (151 MHz, DMSO)  $\delta$  (ppm) 36.2 CH<sub>2</sub> (C8), 115.4 CH C(3/6), 122.3 CH C(1/2), 139.3 C(4/5), 151.1 C(7). **LF<sub>2</sub>**

Table 1 Lists the benzimidazole derivatives names, chemical structures, and abbreviations

Nomenclature	Molecular structure	Abbreviation
(1 <i>H</i> -Benzimidazol-2-yl)methanethiol		<b>LF<sub>1</sub></b>
1-Dodecyl-2-[(dodecylsulfanyl)methyl]-1 <i>H</i> -benzimidazole		<b>LF<sub>2</sub></b>





**Scheme 1** General procedure for the synthesis of **LF<sub>1</sub>** and **LF<sub>2</sub>**. Reagents and conditions: (i) 4 N-HCl, 2 h, reflux; (ii) C<sub>12</sub>H<sub>25</sub>Br, K<sub>2</sub>CO<sub>3</sub>, TBAB, DMF, RT.

was synthesized by direct alkylation reaction following the same procedure described in our previous work.<sup>24</sup> **LF<sub>1</sub>** (0.02 mol) was alkylated using 1-bromododecane (0.03 mol), anhydrous K<sub>2</sub>CO<sub>3</sub> (0.04 mol) and tetra-*n*-butylammonium bromide TBAB as a catalyst in DMF as shown in Scheme 1. The mixture was stirred at room temperature for 48 h. The reaction progress and completion are monitored by thin-layer chromatography. After removing the salts by filtration, the DMF was removed using a rotary evaporator under reduced pressure. The obtained residue was dissolved in dichloromethane and filtered, and the solvent was removed under reduced pressure. The mixture is chromatographed on a silica gel column (eluent: ethyl acetate/cyclohexane (10/90%)). The obtained product **LF<sub>2</sub>** is in a liquid form. Yield: 26%; <sup>1</sup>H NMR (600 MHz, CDCl<sub>3</sub>)  $\delta$  7.99–7.08 (m, 4H), 4.22 (t, 2H), 3.97 (s, 2H), 2.58 (t, 2H), 1.88 (m, 2H), 1.56 (m, 36H), 1.40–1.18 (m, 2H), 0.90 (t, 6H). <sup>13</sup>C NMR (151 MHz, CDCl<sub>3</sub>)  $\delta$  (ppm) 150.8 C(7), 142.3 C(4), 135.5 1C C(5), 122.4, 121.8, 119.5, 109.5 4C C(1, 2, 3 and 6), 44.1, 44.05, 32.8, 31.9, 31.9, 31.7, 29.6, 29.6, 29.6, 29.5, 29.5, 29.5, 29.4, 29.3, 29.2, 29.1, 28.8, 28.7, 28.2, 28.1, 27.0 21C CH<sub>2</sub>, 22.7 2C CH<sub>2</sub>, 14.2 2C CH<sub>3</sub>.

### 2.3. Electrochemical studies

The VoltaLab potentiostat, which was attached to a typical triple-electrode cell, was used for electrochemical studies. A specimen made of carbon steel with a chosen surface area of 1.0 cm<sup>2</sup> served as the working electrode in this configuration. With platinum wire acting as the counter electrode with the same surface area as the working electrode, the reference electrode was a saturated calomel electrode (SCE). Every experiment was finished with fewer than 30 minutes of immersion time to provide a consistent open circuit potential.

**2.3.1. Potentiodynamic polarization studies.** For electrochemical kinetics, polarization curves of the C.S-solution interface are crucial; this technique takes into account the electrochemical interface as the slowest stage of the entire process. VoltaLab PGZ 100 potentiostat, controlled by the VoltaMaster software, performed rigorous electrochemical testing. For all tests, a three-electrode cell equipped with Hg/Hg<sub>2</sub>Cl<sub>2</sub>/KCl (SCE) as the reference electrode, the counter electrode (CE) is a platinum wire with a surface area of 1 cm<sup>2</sup>, and the working electrode (WE) is made of carbon steel with the same surface area. Before the appropriate analysis, the C.S was immersed during 1800 s in each acidic solution and given time to reach a stable open circuit potential. The potentiodynamic polarization records were performed with a sweep rate of 0.5 mV s<sup>-1</sup> between 0.8 V per SCE and 0 V per SCE. The following formula sums up the inhibitory action ( $\eta_{PPD}$ ):

$$\eta_{PPD} (\%) = \frac{i_{cor}^0 - i_{cor}^{inh}}{i_{cor}^0} \times 100 \quad (1)$$

In this context,  $i_{cor}^0$  and  $i_{cor}^{inh}$  represent the corrosion current density values in the absence and presence of the inhibitor, respectively.

### 2.3.2. Electrochemical impedance spectroscopy study.

Electrochemical impedance spectroscopy measurements were made with a VoltaLab PGZ 100 transfer function analyzer with a low-amplitude AC signal (10 mV) at 10 points per decade spanning a frequency range of 100 kHz to 10 mHz. The Nyquist representation was used to create electrochemical impedance diagrams. The data were then analyzed using Z-View software using an analogous electrical circuit. The inhibitory efficiency inferred from the electrochemical impedance spectroscopy was computed using eqn (2).

$$\eta_{SIE} = \frac{R_p - R_p^0}{R_p} \times 100 \quad (2)$$

whereas the values of resistance to polarization in the absence and presence of an inhibitor are denoted by  $R_p^0$  and  $R_p$ , respectively.

We used the results of the blank in the absence and presence of HCl without inhibitor for the effect of concentration and temperature, since we worked under similar conditions in a work already published by our team.<sup>23</sup>

### 2.4. Morphology investigations SEM

The SEM technique was utilized to examine the surfaces. A corrosive solution of molar HCl, with and without an optimal concentration of 0.001 M of inhibitors **LF<sub>1</sub>** and **LF<sub>2</sub>** individually, was employed. The samples were exposed to these harsh environments separately for 24 hours, then delicately removed, rinsed with purified water, dried, and subjected to surface morphological analysis using SEM. The JEOL-JSM-IT-100 model was employed for examining surface morphology. We have used the results of the blank in the absence and presence of HCl without inhibitor since we have worked under similar conditions in a work already published by our team.<sup>23</sup>

### 2.5. AFM analysis

The examination of the deposited film morphology was conducted at  $T = 303$  K through observation using Atomic Force Microscopy (AFM) using Hitachi 5100N. The AFM studies are carried out utilizing an imaging techniques system by immersing the electrode in 1.0 M HCl in the absence and presence of 0.001 M of the two investigated compounds for 24 h.



We have used the results of the blank in the absence and presence of HCl without inhibitor since we have worked under similar conditions in a work already published by our team.<sup>23</sup>

## 2.6. Contact angle

Based on Biolin Scientific's Attension/Theta model, contact angles were determined using the sessile water drop technique. One minute after the liquid was placed on the coated surface, water contact angles (WCA) depending on the profile of the drop attained after equilibrium were determined. Contact angle measurements were employed to analyze the changes in C.S's hydrophobicity when exposed to an acidic environment with two inhibitors, **LF<sub>1</sub>** and **LF<sub>2</sub>**. These assessments were conducted after immersing C.S in the acid solution for 24 hours, both before and after exposure to an optimal inhibitor concentration (0.001 M) of **LF<sub>1</sub>** and **LF<sub>2</sub>**. We have used the results of the blank in the absence and presence of HCl without inhibitor since we have worked under similar conditions in a work already published by our team.<sup>23</sup>

## 2.7. UV-visible spectrum analysis

UV-vis spectroscopy offers a better understanding of the interactions between steel and the inhibitors examined. Using an optimum concentration of 0.001 M benzimidazole derivatives at a temperature of 303 K for 24 hours, the electrolytic solution was subjected to analysis by this method before and after immersion of the C.S samples in a 1.0 M HCl solution. With a spectral range of 0 nm/400 nm, UV-vis spectrophotometer (Jasco type spectrophotometer (series V-730) with a stray light of 0.02% for exceptional absorbance linearity up to 3 abs) was used to record the spectra.

## 2.8. Computational chemical details

Structure, electron distribution, molecule adsorption on metal and oxide surfaces, and studied inhibitory mechanisms are all subjects of investigation. The quantum chemical technique using DFT to examine the effectiveness of inhibitors in relation to their molecular characteristics.<sup>24,25</sup> This study used the DFT/(B3LYP) calculations with base 6-311+G(d,p) performed using Gaussian 09 methods to optimize the benzimidazole derivatives completely. These are looking for molecular descriptors like LUMO ( $E_L$ ) and HOMO ( $E_H$ ) energies. Based on these values, we can calculate energy gap ( $\Delta E = E_L - E_H$ ), absolute electronegativity ( $\chi = \frac{1}{2}(E_H + E_L)$ ), global hardness ( $\eta = -\frac{1}{2}(\Delta E)$ ), softness ( $S = \frac{1}{\eta}$ ), electrophilicity ( $\omega = \frac{\chi^2}{2\eta}$ ), dipole moment ( $\mu$ ), ionization energy ( $I = -E_H$ ), electron affinity ( $A = -E_L$ ), and number of electrons transferred from the inhibitor to the metal Fe(110) ( $\Delta N_{110} = \frac{\Phi - \chi_{inh}}{2\eta_{inh}}$ ) with  $\Phi$  is work function, he energy change during the electronic back-donation process ( $\Delta N_{b-d} = -\frac{\eta_{inh}}{4}$ ).<sup>26-28</sup>

In addition, the vertical ionization potential (VIP) and vertical electron affinity (VEA) are calculated as:<sup>26</sup>

$$IP_v = E_0^+ - E_0 \text{ and } EA_v = E_0^- - E_0 \quad (3)$$

where  $E_0$  is the total energies of the optimized structure,  $E_0^+$  and  $E_0^-$  are the energies of cationic radical and anionic radical at the optimized structure of the neutral species, respectively.

## 2.9. MD simulation details

Molecular dynamics (MD) simulations have become increasingly popular in studying corrosion inhibition in recent years.<sup>29,30</sup> This method helps researchers to understand how inhibitor molecules interact and are adsorbed onto supporting surfaces.<sup>31</sup> In this study, we focused on a system consisting of 9Cl<sup>-</sup>, 9H<sub>3</sub>O<sup>+</sup>, and 481H<sub>2</sub>O molecules, and successfully built a monomer of the selected molecule during each simulation.

We used the Forceite technique in Materials Studio 8 software to run the molecular dynamics simulations, with cell characteristics of 30 Å vacuums, periodic boundary conditions, a unit cell size of (27.30 × 27.30 × 40.13 Å<sup>3</sup>), and a (11 × 11) unit cell. Prior to the simulation, the **LF<sub>1</sub>** and **LF<sub>2</sub>** structures were pre-optimized using the GGA (Generalized Gradient Approximation) & DNP (Dual Numerical Polarization) functions.

The COMPASS force field was used to run the simulation, with a time interval of 1.0 fs and a duration of 2000 ps. We used the *NVT* ensemble with an Andersen thermostat to run all simulations at a temperature of 303 K conditions, (27.30 × 27.30 × 40.13 Å<sup>3</sup>) unit cell size, and (11 × 11) unit cell. The **LF<sub>1</sub>** and **LF<sub>2</sub>** structures are pre-optimized through the use of the GGA (Generalized Gradient Approximation) & DNP (Dual Numerical Polarization) functions. The simulation was run using the COMPASS force field, with a time interval of 1.0 fs and a duration of 2000 ps. The *NVT* ensemble with an Andersen thermostat was used to run all simulations at a temperature of 303 K.<sup>32</sup>

## 3. Results and discussions

### 3.1. Potentiodynamic polarization measurements

Fig. 1 displays potentiodynamic polarization plots for carbon steel at 303 K in 1.0 M HCl solution, both with and without different **LF<sub>1</sub>** and **LF<sub>2</sub>** concentrations.

Eqn (4)<sup>33</sup> was used in the Tafel extrapolation method to determine the corrosion inhibition efficiency ( $\eta_{PPD}$ ) because every reaction exhibits Tafel behavior. The corrosion current density obtained in the absence of corrosion inhibitors  $i_{corr}^0$  and the corrosion current density observed in the presence of corrosion inhibitors  $i_{corr}^{inh}$  are the two variables in this equation. Table 2 displays the electrochemical corrosion parameters that were determined.

$$\eta_{PPD} (\%) = \frac{i_{corr}^0 - i_{corr}^{inh}}{i_{corr}^0} \times 100 \quad (4)$$

The data presented in Table 2 reveal that the introduction of **LF<sub>1</sub>** and **LF<sub>2</sub>** inhibitors into the corrosive medium results in a decrease in both anodic and cathodic current densities.<sup>34</sup> This implies that both anodic metal attack and cathodic hydrogen

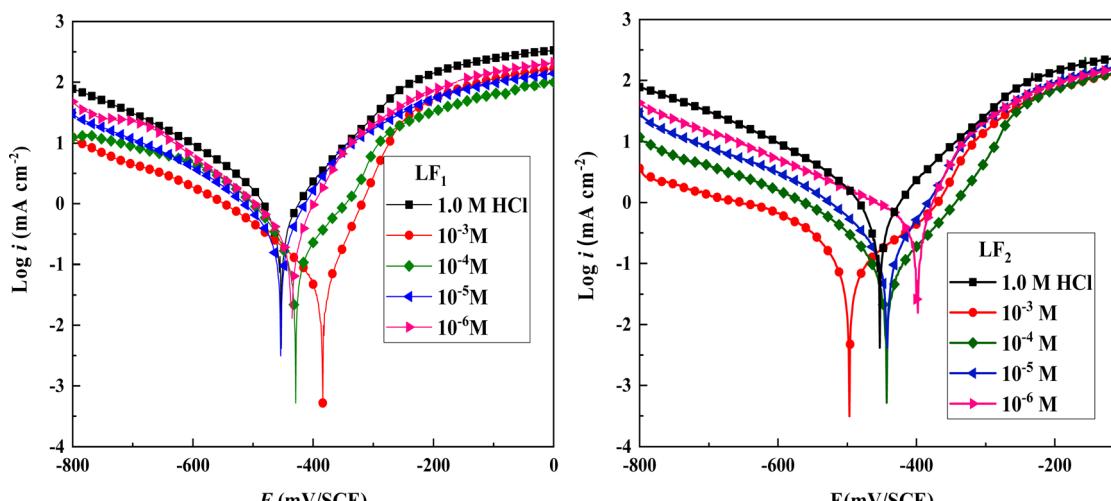


Fig. 1 Potentiodynamic polarization graphs of C.S at 303 K, with and without different **LF**<sub>1</sub> & **LF**<sub>2</sub> concentrations, submerged in 1.0 M HCl solution.

evolution reactions are reduced following the addition of **LF**<sub>1</sub> and **LF**<sub>2</sub> inhibitors to the acid solution. The corrosion-inhibiting effect becomes increasingly pronounced with increasing inhibitor concentration,<sup>35</sup> from  $1 \times 10^{-6}$  M to  $1 \times 10^{-3}$  M. Fig. 1 highlights the fact that the cathodic current–potential curves are almost parallel, suggesting that the addition of the inhibitors **LF**<sub>1</sub> and **LF**<sub>2</sub> to the 1.0 M HCl solution does not alter the mechanism of the hydrogen evolution reaction. The reduction of H<sup>+</sup> ions on the surface of carbon steel occurs mainly *via* a charge transfer mechanism.<sup>36–38</sup> However, the  $\beta_c$  value changes with the benzimidazole addition, indicating that both inhibitors influence the rate at which hydrogen is evolved, which suggests that benzimidazoles powerfully inhibit the corrosion process of C.S, and its ability as a corrosion inhibitor is enhanced as its concentration is increased. The suppression of the cathodic process can be due to the formation of a protective film of benzimidazoles at the metal/solution interface.<sup>36,39,40</sup>

It can be seen, that compared to the blank solution, the value of  $\beta_a$  is markedly changed in the presence of benzimidazoles, which suggests that benzimidazoles can affect the kinetics of the anodic process. Given their several pairs of free electrons, nitrogen, and sulfur atoms in the benzimidazole molecules participate in the formation of the iron–benzimidazole

complex. This interaction between iron and the benzimidazole molecule alters the mechanism of iron dissolution.<sup>36,41,42</sup>

As depicted in Table 2, introducing **LF**<sub>1</sub> into the acidic solution results in a shift of the corrosion potential towards more positive values. This suggests that the inhibitor functions primarily as a mixed-type corrosion inhibitor, with a predominantly anodic effect,<sup>43</sup> according to the polarization measurement data. Conversely, the addition of **LF**<sub>2</sub> to the acidic solution shifts the corrosion potential towards more negative values. **LF**<sub>2</sub> exhibits mixed-type corrosion inhibition, with a predominant cathodic action.<sup>44–46</sup>

### 3.2. Electrochemical impedance spectroscopy study

Electrochemical impedance spectroscopy tests were conducted to learn more about the electrochemical processes occurring across the electrode/solution interface and to examine these properties in the presence and absence of compounds containing benzimidazole derivatives ( $1 \times 10^{-3}$  M,  $1 \times 10^{-4}$  M,  $1 \times 10^{-5}$  M, and  $1 \times 10^{-6}$  M), which are present in the 1.0 M HCl solution. Thanks for sharing the information. We'll take a look at the Nyquist charts in Fig. 2 to better understand the systems.

Nyquist plots and Bode curves obtained for compounds in the **LF**<sub>2</sub>-1.0 M HCl and C.S-**LF**<sub>1</sub> systems at open-circuit-potential

Table 2 Polarization parameters for various concentrations of **LF**<sub>1</sub> and **LF**<sub>2</sub>

Solution	Conc.	$-E_{corr}$ /SCE (mV)	$i_{corr}$ ( $\mu\text{A cm}^{-2}$ )	$\beta_a$ (mV dec <sup>-1</sup> )	$-\beta_c$ (mV dec <sup>-1</sup> )	$\eta_{PPD}$ (%)
<b>LF</b> <sub>1</sub>	1 M	$456.3 \pm 6.0$	$1104.1 \pm 5.0$	112.8	155.4	—
	$10^{-3}$	$385.7 \pm 3.2$	$130.7 \pm 1.2$	66.4	183.1	88.1
	$10^{-4}$	$427.6 \pm 6.2$	$234.0 \pm 2.5$	104.8	115.6	78.8
	$10^{-5}$	$454.2 \pm 5.2$	$330.7 \pm 3.5$	82.6	125.1	70.0
	$10^{-6}$	$437.0 \pm 7.2$	$407.5 \pm 5.1$	75.7	139.5	63.1
	$10^{-3}$	$497.9 \pm 4.1$	$50.4 \pm 0.3$	89.1	61.7	95.4
<b>LF</b> <sub>2</sub>	$10^{-4}$	$441.7 \pm 3.2$	$85.5 \pm 3.2$	79.4	109.9	92.2
	$10^{-5}$	$441.9 \pm 3.2$	$214.1 \pm 2.4$	70.6	125.3	80.6
	$10^{-6}$	$400.0 \pm 3.2$	$300.3 \pm 3.1$	43.2	99.7	72.8



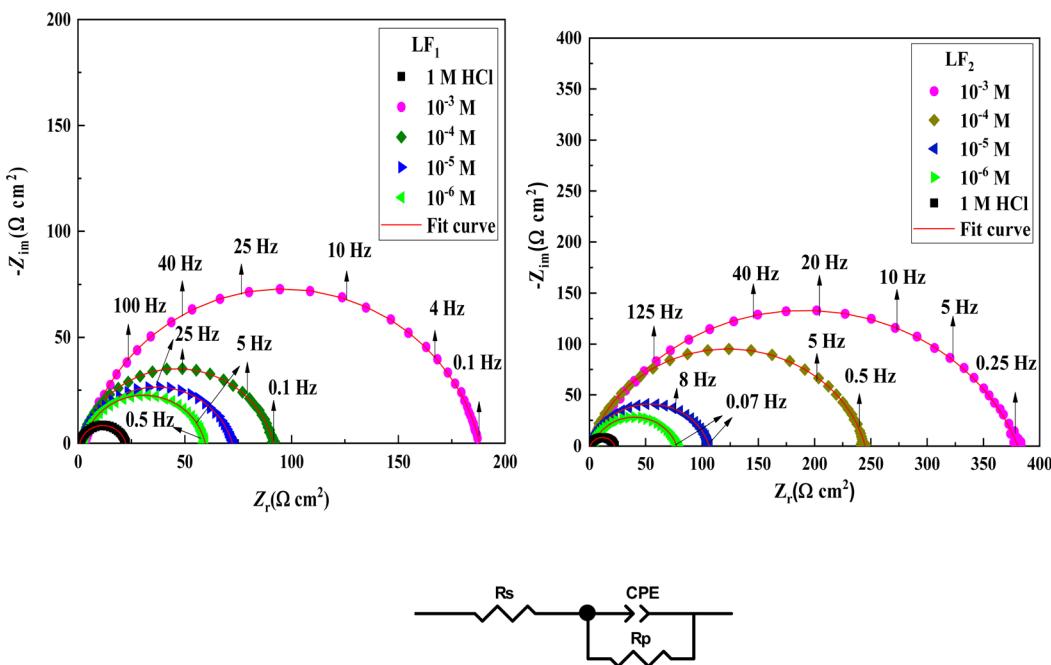


Fig. 2 Impedance spectroscopy Nyquist plots in a 1.0 M HCl solution at various **LF**<sub>1</sub> and **LF**<sub>2</sub> concentrations at 303 K, along with the corresponding equivalent circuit that agrees with the experimental impedance data.

are displayed in Fig. 2 and 3. Table 3 provides the corresponding fitted parameters. It is clear from the acquired spectra that a capacitive loop was present. This loop's diameter increased significantly upon the addition of chemicals **LF**<sub>1</sub> and **LF**<sub>2</sub>, peaking at a concentration of 10<sup>-3</sup> M for each molecule. Furthermore, as reported in the context of frequency dispersion, these plots show intact semicircles.<sup>47</sup> R<sub>p</sub> pointed out that it is found that the resistance to polarization rises with the use of inhibitors. An increase in the inhibitors' effectiveness is indicated by the semicircle's size growing in proportion to the **LF**<sub>1</sub>

and **LF**<sub>2</sub> chemical concentrations. These findings suggest that a charge transfer mechanism controls the corrosion of C.S in hydrochloric solution, and that adsorption of **LF**<sub>1</sub> and **LF**<sub>2</sub> compounds to the C.S surface prevents corrosion.

An analogous electrochemical circuit with a solution resistor ( $R_s$ ), a resistance of polarization ( $R_p$ ), and a constant phase element (CPE) is applied to model impedance behavior frequently. Table 3 presents the computed values of the pertinent electrochemical parameters, which are  $R_s$ ,  $R_p$ , and  $CPE_{dl}$ . Furthermore, the correctness of the simulated data was

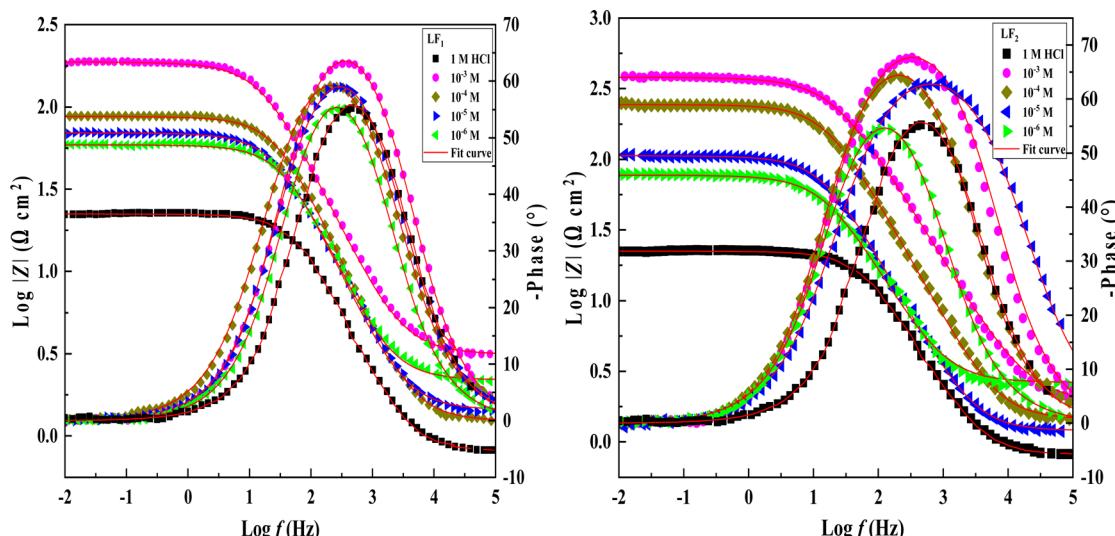


Fig. 3 Bode plots & phase angle spectral representations of Bode for the impedance spectroscopy system in a 1.0 M HCl solution, both with and without different **LF**<sub>1</sub> & **LF**<sub>2</sub> concentrations at 303 K.



Table 3 Electrochemical impedance values in the presence and absence of **LF<sub>1</sub>** and **LF<sub>2</sub>**

Medium	Conc. (M)	$R_s$ ( $\Omega$ cm <sup>2</sup> )	$R_p$ ( $\Omega$ cm <sup>2</sup> )	$10^6 \times Q$ ( $\mu\text{F s}^{n-1}$ cm <sup>-2</sup> )	$n$	$C_{dl}$ ( $\mu\text{F cm}^{-2}$ )	$\chi^2$	$\eta_{SIE}$ (%)
<b>HCl</b>	1 M	0.83 ± 0.01	21.57 ± 0.56	293.9 ± 2.35	0.845 ± 0.003	116.2	0.002	—
	$10^{-3}$	3.94 ± 0.03	184.9 ± 1.54	83.2 ± 0.90	0.862 ± 0.005	42.62	0.004	88.3
	$10^{-4}$	1.25 ± 0.01	90.4 ± 0.83	125.0 ± 1.32	0.857 ± 0.006	59.17	0.005	76.2
	$10^{-5}$	1.55 ± 0.01	70.9 ± 0.61	163.0 ± 1.51	0.851 ± 0.004	74.64	0.005	69.6
	$10^{-6}$	2.35 ± 0.02	57.4 ± 0.49	175.0 ± 1.83	0.849 ± 0.001	77.2	0.004	62.5
<b>LF<sub>2</sub></b>	$10^{-3}$	1.70 ± 0.02	378.7 ± 2.77	29.2 ± 0.26	0.851 ± 0.002	13.26	0.004	94.3
	$10^{-4}$	1.53 ± 0.01	243.5 ± 2.09	59.9 ± 0.56	0.861 ± 0.005	30.26	0.005	91.1
	$10^{-5}$	1.01 ± 0.01	106.0 ± 0.96	165.0 ± 1.58	0.843 ± 0.001	77.66	0.007	79.7
	$10^{-6}$	1.28 ± 0.01	76.5 ± 0.58	230.5 ± 2.34	0.831 ± 0.002	101.4	0.006	71.9

evaluated using chi-square; small chi-square values ( $10^{-3}$ ) (Table 3) show that the simulated and experimental data agree closely.

Impedance spectrum depression is commonly attributed to frequency dispersion. This frequency dispersion is due to inhomogeneities in the electrode surface (formation of corrosion products, roughness, presence of impurities, variations in the thickness or composition of a film or coating on the metal surface, or inhibitor adsorption), which induce a change in the electrode's active surface. These surface inhomogeneities are accounted for by a constant-phase element CPE ( $Q, n$ ), via the coefficient  $n$  (between 0 and 1). The impedance of such an element is given by the following equation:<sup>48</sup>

$$Z_{CPE} = Q^{-1}(i\omega)^{-1} \quad (5)$$

In this equation, the imaginary component, angular frequency, deviation indicator, and CPE constant are denoted as  $Q$ ,  $\omega$ ,  $n$ , and CPE, respectively. The value of  $n$  is linked to the uniformity of the electrode surface. Additionally, the following formula is used to estimate the values of the double/layer capacitance ( $C_{dl}$ ):

$$C_{dl} = (Q \times R_p^{1-n})^{1/n} \quad (6)$$

$\eta_{SIE}$  was determined according to the following expression:

$$\eta_{SIE} = \frac{R_p^0 - R_p}{R_p} \times 100 \quad (7)$$

where  $R_p^0$  and  $R_p$  denote the charge transfer resistance in the non-inhibited and inhibited medium, respectively.

Table 3 shows the parameters obtained from the Electrochemical Impedance Spectroscopy studies. It was observed that while  $C_{dl}$  values show an inverse relationship,  $R_p$  increases proportionally with inhibitor concentration. This phenomenon is probably due to the adsorption of inhibitor compounds, which increased the surface coverage of the C.S. The increase in the thickness of the double layer, caused by these organic compounds replacing the water molecules at the C.S interface, is linked to the decrease in  $C_{dl}$ . Due to the irregularity of the metal surface induced by the development of porous layers, the deviation of  $C_{dl}$  from perfect capacitive behaviour can be associated with the slight divergence of the  $n$  value from unity.<sup>49,50</sup>

The chi-squared ( $\chi^2$ ) was utilized to appraise the precision of the fitting outcomes, the small chi-squared values (Table 3) acquired for all the outcomes show that the fitted results have

a great concurrence with the experimental findings. For **LF<sub>2</sub>** at the optimal concentration ( $10^{-3}$  M), the maximum value of  $R_p$  ( $378.7 \Omega$  cm<sup>2</sup>) and the minimum value of  $C_{dl}$  ( $13.26 \mu\text{F s}^{n-1}$  cm<sup>-2</sup>) were discovered. This finding demonstrates that **LF<sub>2</sub>** exhibits stronger inhibitory performance than **LF<sub>1</sub>**. For **LF<sub>1</sub>**, the obtained value of  $\eta_{SIE}$  is 88.3%. The inhibitory efficacy of the compound rises with the addition of a second benzimidazole segment, reaching a value of 94.3% for **LF<sub>2</sub>**. The interaction between the many N, S, and  $\pi$ -electron active sites and the open orbitals of the iron atoms explains this enhancement. Furthermore, **LF<sub>2</sub>**'s total contact area is greater than **LF<sub>1</sub>**'s, leading to a more noticeable interaction between the steel surface.<sup>51</sup>

The acquired results (refer to Table 2) are in good agreement with the potentiodynamic polarization measurements.

### 3.3. Adsorption isotherm

Various types of isotherm models were applied to better understand how the **LF<sub>1</sub>** and **LF<sub>2</sub>** studies interact with the C.S surface (Fig. S1 in the ESI†). The correlation coefficient ( $R^2$ ) was found to be the main parameter fitted to the experimental data. Surface coverage rates ( $\theta$ ) were determined using the following formula for different quantities:

$$\theta = \frac{\eta_{SIE}}{100} = \left( \frac{R_p^0 - R_p}{R_p} \right) \quad (8)$$

The  $C_{inh}/\theta$  vs.  $C_{inh}$  plot gives a straight line with a slope near the unit (Fig. S1†), and also, a linear correlation coefficient near 1, which shows the adsorption of **LF<sub>1</sub>** and **LF<sub>2</sub>** obeys Langmuir isotherm that expressed according to eqn (9):<sup>51,52</sup>

$$C_{inh} \times \theta^{-1} = K^{-1} + C_{inh} \quad (9)$$

where  $K$  which is connected to the Gibbs free energy, stands for the adsorption equilibrium ( $\Delta G_{ads}$ ) using the formula below:<sup>53</sup>

Table 4 Thermodynamic adsorption characteristics of **LF<sub>1</sub>** & **LF<sub>2</sub>** on the C.S surface in 1.0 M HCl medium

	$R^2$	Slope	$K$ (L mol <sup>-1</sup> )	$\Delta G_{ads}$ (kJ mol <sup>-1</sup> )
<b>LF<sub>1</sub></b>	0.999	1.12	$1.4261 \times 10^5$	-40.0
<b>LF<sub>2</sub></b>	0.999	1.05	$4.9907 \times 10^5$	-43.1



Table 5 Potentiodynamic polarization parameters for C.S without and with  $10^{-3}$  M of  $\text{LF}_1$  &  $\text{LF}_2$  at diverse  $T$ 

Solution	$T$ (K)	$-E_{\text{corr}}$ (mV SCE)	$i_{\text{corr}}$ ( $\mu\text{A cm}^{-2}$ )	$\beta_a$ (mV dec $^{-1}$ )	$-\beta_c$ (mV dec $^{-1}$ )	$\eta_{\text{pp}}$ (%)
1.0 M HCl	303	$456.3 \pm 6.0$	$1104.1 \pm 5.0$	112.8	155.4	—
	313	$423.5 \pm 9.0$	$1477.4 \pm 8.0$	91.3	131.3	—
	323	$436.3 \pm 7.0$	$2254.0 \pm 10.0$	91.4	117.8	—
	333	$433.3 \pm 5.0$	$3944.9 \pm 12.0$	103.9	134.6	—
	LF <sub>1</sub>	$385.7 \pm 3.2$	$130.7 \pm 1.2$	66.4	183.1	88.1
	313	$432.6 \pm 6.2$	$292.9 \pm 3.0$	71.4	137.7	80.1
	323	$420.9 \pm 5.3$	$516.1 \pm 5.1$	53.1	146.7	77.1
	333	$407.9 \pm 6.0$	$1258.7 \pm 10.9$	49.2	209.1	68.1
LF <sub>2</sub>	303	$497.9 \pm 4.1$	$50.4 \pm 0.3$	89.1	61.7	95.4
	313	$490.9 \pm 7.5$	$146.2 \pm 1.5$	85.8	134.0	90.1
	323	$499.7 \pm 1.7$	$365.1 \pm 3.7$	71.6	117.7	83.8
	333	$451.01 \pm 5.8$	$1138.2 \pm 12.0$	84.9	116.6	71.1

$$\Delta G_{\text{ads}} = -RT \times \ln(1 \times 55.5 \times K) \quad (10)$$

where the water content in the solution is denoted by  $(1 \times 55.5)$ .

The Langmuir isotherm adsorption parameters (Table 4) have a regression coefficient  $R^2$  of 1 for  $\text{LF}_1$  & for  $\text{LF}_2$ . It is

generally acknowledged that a result above  $-20 \text{ kJ mol}^{-1}$  shows physical adsorption and that chemical adsorption occurs below  $-40 \text{ kJ mol}^{-1}$ . The interaction between inhibitors and metals is thought to be mixed adsorption, comprising both chemical and physical adsorption if the value falls between these two criteria.

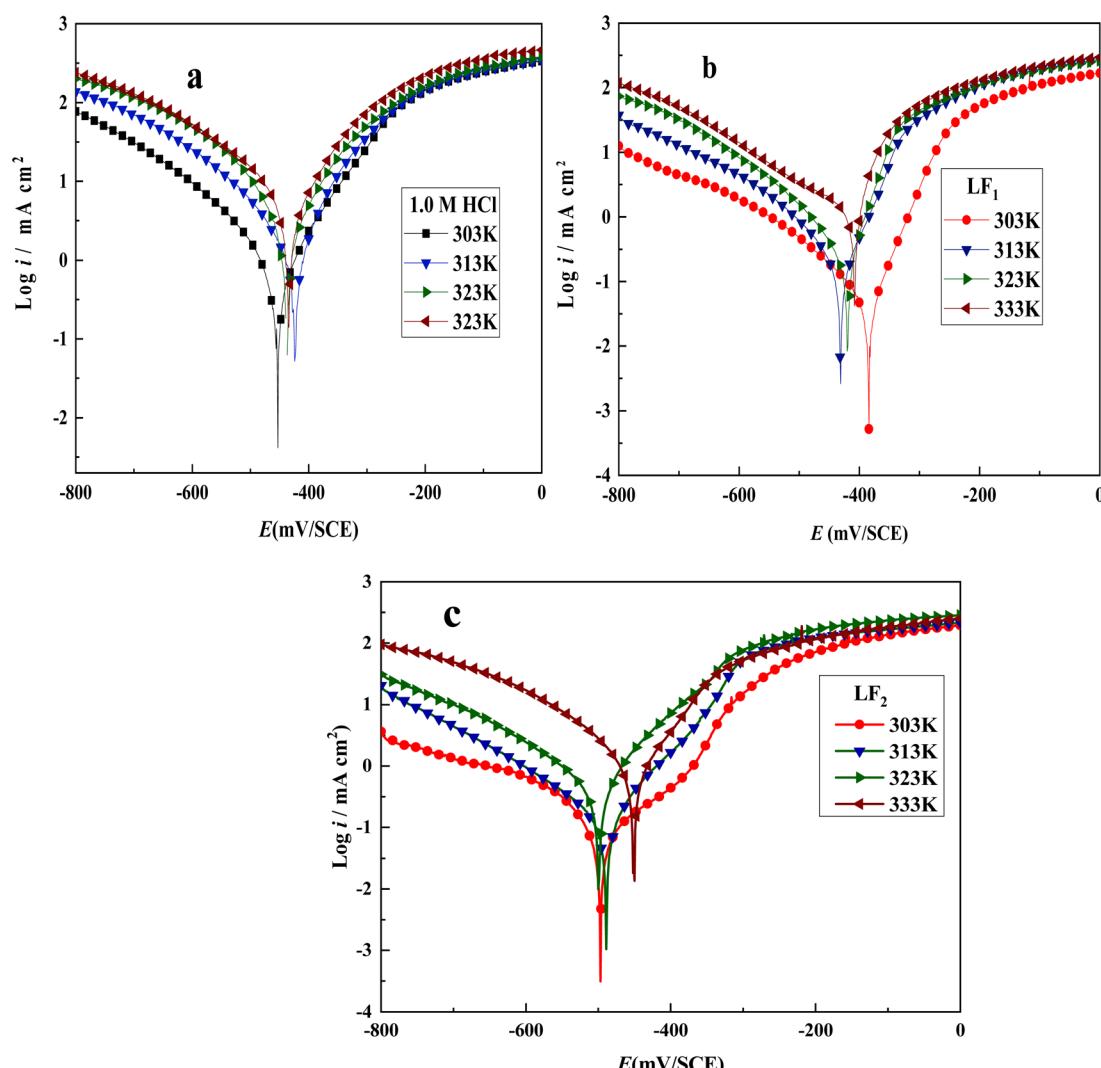


Fig. 4 Tafel plots showing C.S in 1.0 M HCl solution with and without inhibitors at  $10^{-3}$  M concentrations at varying temperatures: (a) without inhibitor, (b) with  $\text{LF}_1$ , and (c) with  $\text{LF}_2$ .



$\Delta G_{\text{ads}}$  calculated for **LF<sub>1</sub>** and **LF<sub>2</sub>** at the temperature 303 K, show that the shaping of adsorbed film on the surface of the C.S is mostly the product of the inhibitory mechanism's reinforcement of the chemical adsorption process.<sup>54</sup> Although the  $\Delta G_{\text{ads}}$  values strongly indicate that chemisorption is the primary adsorption mode, it is not possible to completely discount the influence of van der Waals interactions (physical adsorption). These interactions might be significant during the initial adsorption stages or contribute to the adsorbed layer's overall stability. Consequently, the adsorption process on the C.S surface is considered a combination of chemisorption and physisorption, with chemisorption being the predominant interaction based on the estimated thermodynamic outcomes.

### 3.4. Effect of temperature

The temperature of the corrosive environment influences the interaction between the C.S surfaces and the inhibitors, as well as the corrosion rate of C.S in an acid solution. To examine this effect, polarization analyses were carried out on C.S in a 1.0 M HCl solution, in the presence and absence of inhibitors at a concentration of  $10^{-3}$  M, over a temperature range from 303 to 333 K. The results are shown in Table 5 and Fig. 4.

As corrosion current density rises in both inhibited and uninhibited solutions, both compounds function as temperature-sensitive inhibitors, as shown by the results in Fig. 4 and Table 5. Moreover, as the temperature of the corrosive solutions rises, the inhibitory efficacy of both compounds falls. Therefore, inhibitory performance is decreased at higher temperatures. This observation is most likely the result of inhibitor compounds' declining adsorption capacity as temperature rises.

Eqn (11) and (12) were used to compute thermodynamic activation characteristics, such as activation energies ( $E_{\text{a}}$ ), enthalpies ( $\Delta H_{\text{a}}$ ), and activation entropies ( $\Delta S_{\text{a}}$ ), which were then combined and displayed in Table 6.

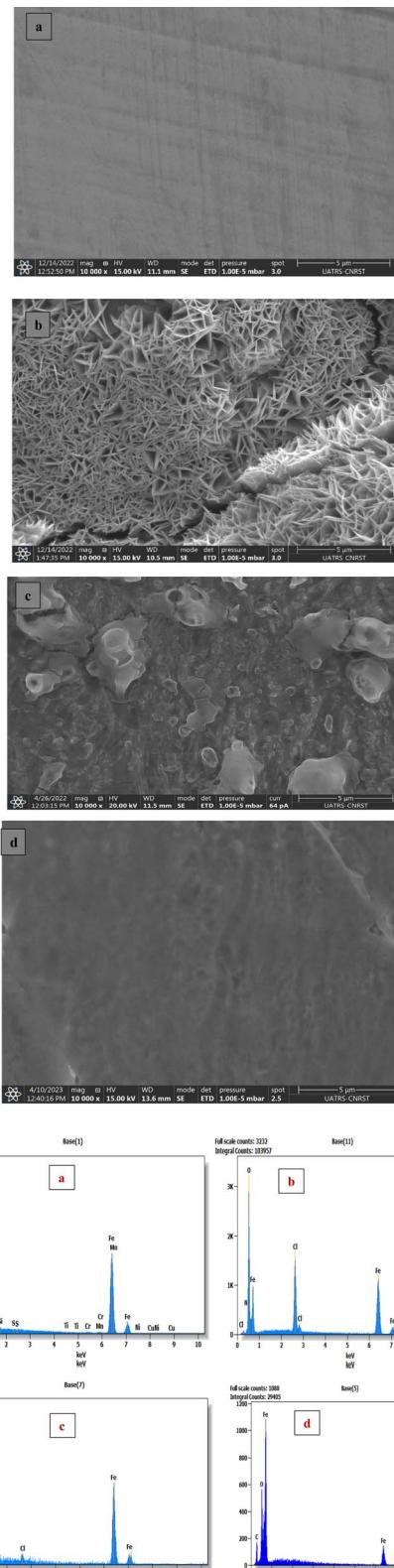
$$i_{\text{corr}} = A e^{\left(\frac{-E_{\text{a}}}{RT}\right)} \quad (11)$$

$$i_{\text{corr}} = \frac{RT}{Nh} \exp\left(\frac{\Delta S_{\text{a}}}{R}\right) \exp\left(\frac{-\Delta H_{\text{a}}}{RT}\right) \quad (12)$$

where  $R$ ,  $A$ ,  $N$ , and  $h$  indicate the gas constant, pre-exponential constant, Avogadro number, and Planck constant, respectively. To determine  $E_{\text{a}}$ , various values of  $\ln(i_{\text{corr}})$  as a function of  $1/T$  are listed in Fig. S2.<sup>†</sup> The Arrhenius transition equation (eqn (12)) was utilized to calculate the enthalpy ( $\Delta H_{\text{a}}$ ) and activation entropy ( $\Delta S_{\text{a}}$ ). A right line is produced by comparing  $\ln\left(\frac{i_{\text{corr}}}{T}\right)$  to

**Table 6** Activation parameters of 1.0 M HCl C.S without and with the addition of  $1 \times 10^{-3}$  M **LF<sub>1</sub>** & **LF<sub>2</sub>**

	$R^2$	$E_{\text{a}}$ (kJ mol <sup>-1</sup> )	$\Delta H_{\text{a}}$ (kJ mol <sup>-1</sup> )	$\Delta S_{\text{a}}$ (J mol <sup>-1</sup> K <sup>-1</sup> )
1.0 M HCl	0.9670	35.40	32.77	-79.2
<b>LF<sub>1</sub></b>	0.9860	61.50	59.00	-9.88
<b>LF<sub>2</sub></b>	0.9960	85.90	83.30	62.50



**Fig. 5** SEM/EDX of C.S only (a), with 1.0 M HCl (b), and with  $1 \times 10^{-3}$  M of **LF<sub>1</sub>** (c) & **LF<sub>2</sub>** (d) after 24 h of immersion.



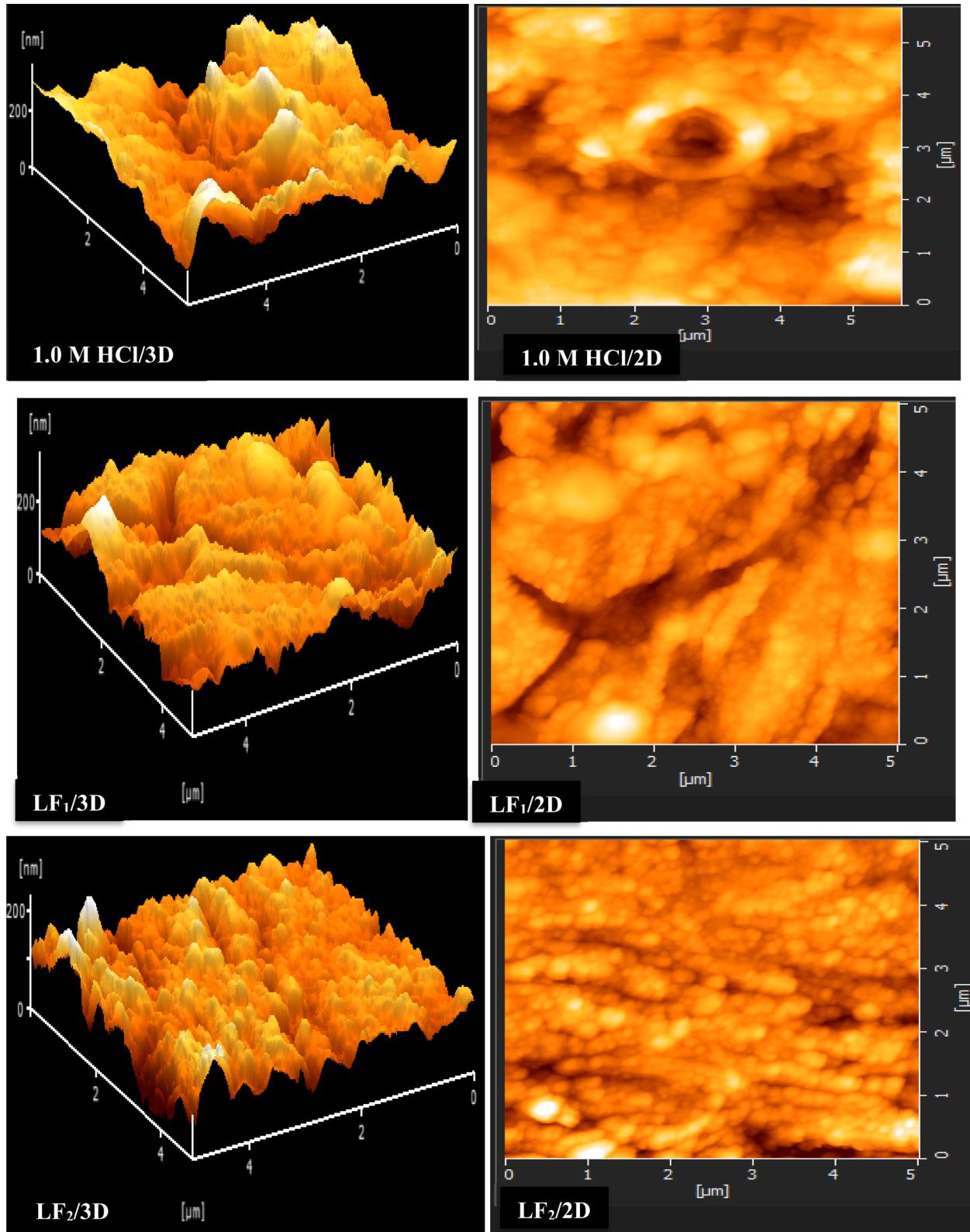


Fig. 6 AFM images (2D and 3D) were captured for C.S specimens both in the non-existent and in the existence of  $1 \times 10^{-3}$  M concentrations of LF<sub>1</sub> & LF<sub>2</sub>.

$(T^{-1})$  (Fig. S2†). Upon seeing the point where this line intersects the  $\ln\left(\frac{i_{\text{corr}}}{T}\right)$  axis, we get  $\Delta S_a$ , which has a slope of  $-\Delta H_a \times R^{-1}$ .

The inhibition process was accelerated by the inhibitors under investigation, which raised the energy barrier ( $E_a$ ) of the inhibition solution in comparison to the blank. This can be explained by these molecules gradually adhering to the C.S surface to form a protective layer. The positive indication of enthalpy ( $\Delta H_a$ ) indicates that the C.S dissolving process is endothermic ( $\Delta H_a$ ).<sup>55</sup> The activation entropy  $\Delta S_a$  increases following the addition of **LF<sub>1</sub>** and **LF<sub>2</sub>**, suggesting an increase in molecular disorder in the system.<sup>56,57</sup>

### 3.5. Surface characterizations

**3.5.1. SEM-EDX analysis.** Scanning electron micrographs (SEM) of the C.S surface immersed in 1.0 M HCl solution, with and without the addition of  $1 \times 10^{-3}$  M **LF<sub>1</sub>** and **LF<sub>2</sub>** for 24 h at 303 K, are shown in Fig. 5a–c. Before immersion in 1.0 M HCl, the C.S substrate has a smooth surface with marks resulting from pre-treatment (Fig. 5a). After immersion in 1.0 M HCl alone (Fig. 5b), the C.S substrate is highly corroded, showing numerous cavities due to the aggressive attack of chloride ions. However, the addition of  $1 \times 10^{-3}$  M of the inhibitors **LF<sub>1</sub>** and **LF<sub>2</sub>** (Fig. 5c and d) reduces surface damage, with fewer cavities observed, suggesting that **LF<sub>1</sub>** and **LF<sub>2</sub>** offer significant protection to the C.S surface against the aggressive electrolyte.

Severe damage to the sample's surface has been caused by rapid corrosion that was observed in a 1.0 M HCl acid solution without **LF<sub>1</sub>** or **LF<sub>2</sub>**, as shown in the C.S picture (Fig. 5b). However, the surface appears to be covered with plate-like patterns (Fig. 5c and d) when **LF<sub>1</sub>** or **LF<sub>2</sub>** is present at a concentration of  $1 \times 10^{-3}$  M, indicating the existence of organic molecules (Fig. 5c and d). This observation suggests

that the prevention of corrosion occurs when inhibiting molecules produce a deposit that creates a layer on the C.S surface.

**3.5.2. AFM characterization.** Fig. 6 shows the 2D and 3D images that were produced using AFM analysis of steel samples that were submerged in a solution of 1.0 M HCl acid solution for 24 hours at 303 K, both in the non-existent of any inhibitor (blank) & with  $1 \times 10^{-3}$  M concentrations of **LF<sub>1</sub>** and **LF<sub>2</sub>** compounds. AFM is renowned for its capacity to evaluate sample surface roughness and produce high-resolution images.<sup>58</sup> Fig. 6 shows the corresponding height profiles. Rough micrographs were seen in the case of the uninhibited C.S sample (Fig. 7), suggesting that metals are more susceptible to corrosion.

The addition of **LF<sub>1</sub>** and **LF<sub>2</sub>** produced a C.S surface that was noticeably uniformly smooth and deepened the corrosion grooves significantly. These findings imply that these organic substances are useful in reducing the rate at which steel samples corrode in 1.0 M HCl.

**3.5.3. Contact angle.** Contact angle evaluations were used to assess the variation in C.S's hydrophobicity in an acidic media, as shown in Fig. 8. The hydrophilic character of the C.S surface is demonstrated by the contact angle of  $25.35^\circ$  when the inhibitors **LF<sub>1</sub>** & **LF<sub>2</sub>** are not present. On the other hand, the contact angle rises when these chemicals are present at a concentration of  $1 \times 10^{-3}$  M. Specifically, the contact angle rises to  $46.66^\circ$  and  $64.09^\circ$ , respectively, in the presence of **LF<sub>1</sub>** & **LF<sub>2</sub>** at a concentration of  $1 \times 10^{-3}$  M. This enhanced hydrophobicity is indicated by the increased contact angle with **LF<sub>1</sub>** and **LF<sub>2</sub>**, which implies a decrease in the wettability of the steel surface. The adsorption and film development of **LF<sub>1</sub>** & **LF<sub>2</sub>** on the steel substrate is responsible for this modification.<sup>59</sup> The fact that these values are below  $90^\circ$  suggests that the film is hydrophilic. The organic coating that forms on the C.S surface is less hydrophilic than in the case of the free acid solution, as

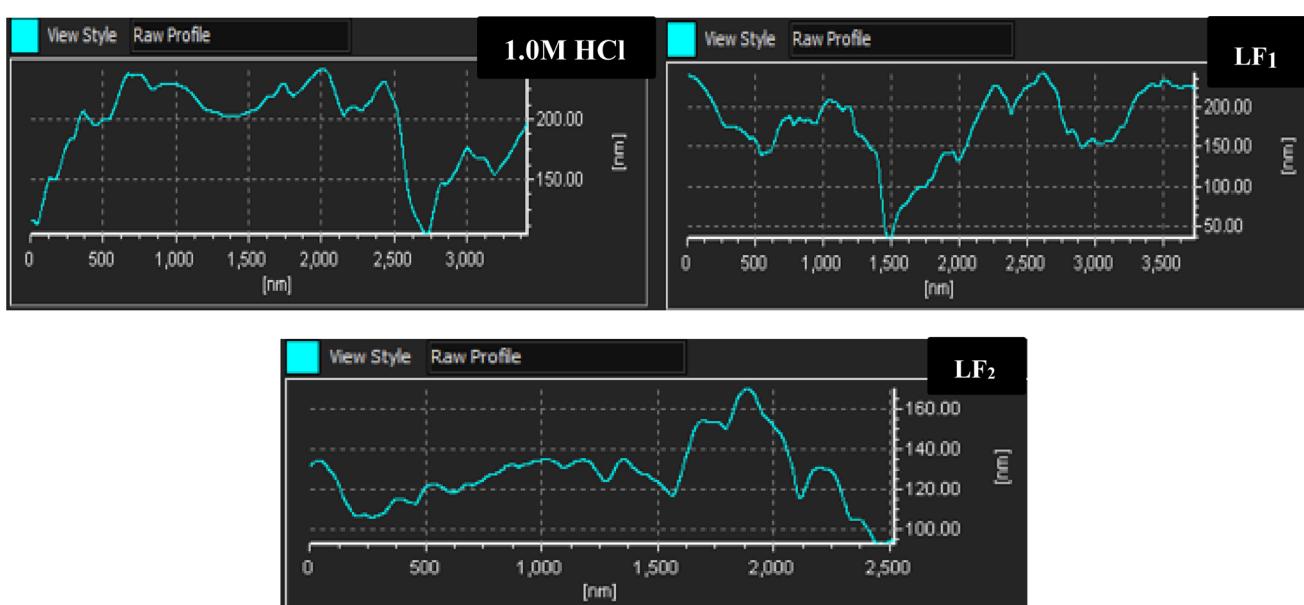


Fig. 7 Displays the relative-height profiles extracted from 3D/2D AFM images at a temperature of 303 K.



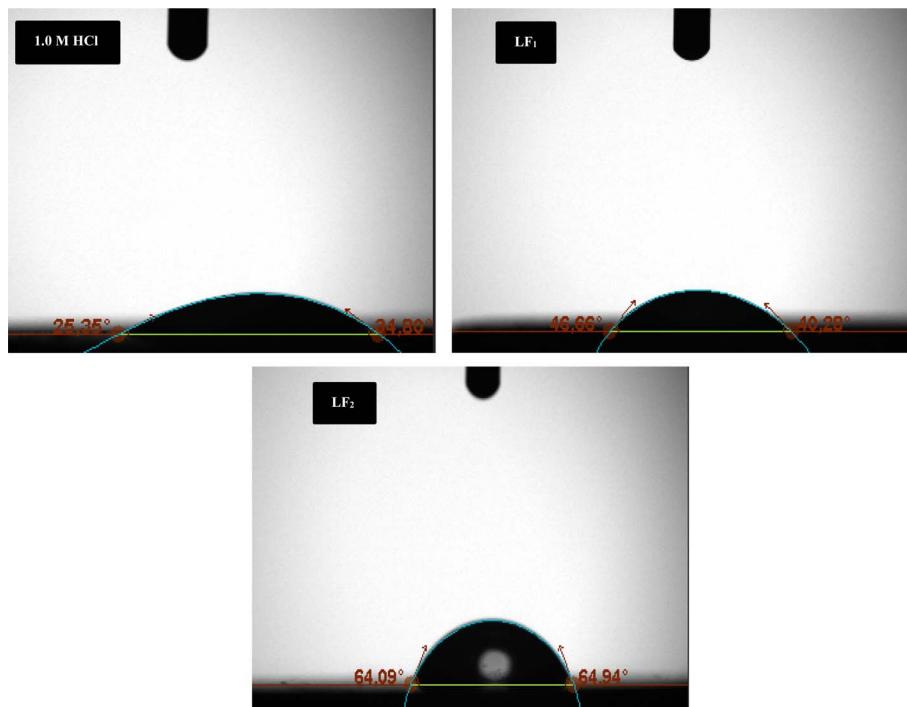


Fig. 8 Displays the contact angle measurements of C.S submerged in 1.0 M HCl for 24 h, both without (blank) & with the existence of  $1 \times 10^{-3}$  M of LF<sub>1</sub> and LF<sub>2</sub>.

evidenced by the increase in contact angle seen in the case of LF<sub>1</sub> and LF<sub>2</sub> addition. These findings clearly demonstrate that adding LF<sub>1</sub> and LF<sub>2</sub> to the corrosive media causes an adsorbed organic coating to develop on the steel surface, thereby preventing C.S acid corrosion.

**3.5.4. UV-visible.** One useful method for identifying complex ions is to employ monochromatic light absorption, which is directly proportional to the absorber species' concentration. Within regular analysis, a straightforward yet conventional approach based on UV-visible absorption turns out to be more sensible than direct spectrophotometric detection.

Variations in the location of the highest absorbance and/or magnitude of the absorbance indicate the formation of a complex between two species in solution.

The UV-visible (UV-visible) spectra in the present investigation indicate a change in wavelength ( $\lambda$ ) for the corrosive solution of 1.0 M HCl containing only the inhibitors LF<sub>1</sub> and LF<sub>2</sub> in comparison to the solution of 1.0 M HCl containing both inhibitors in the presence of C.S (Fig. 9). This demonstrates how the complexation reaction helps the two inhibitors, LF<sub>1</sub> and LF<sub>2</sub>, decrease the solubility of ferric ions Fe<sup>2+</sup> in the corrosive solution.<sup>6,60</sup>

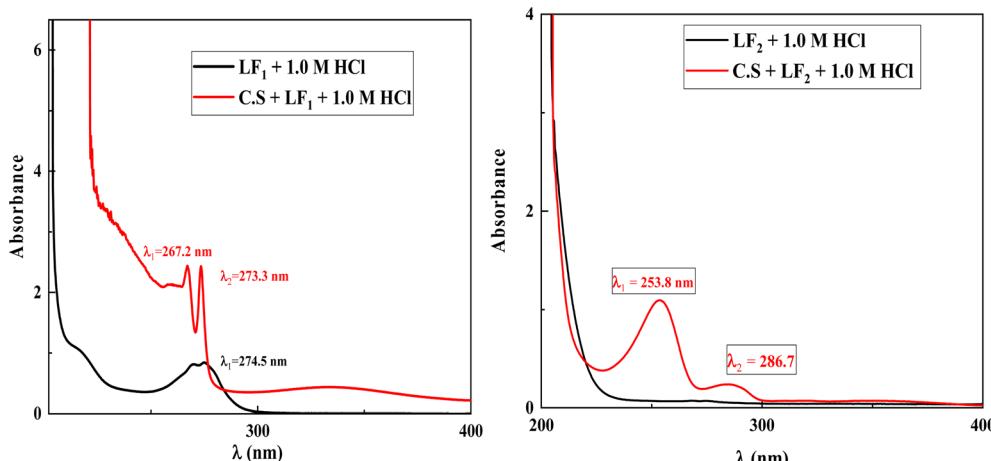


Fig. 9 UV-visible spectrums of benzimidazole derivatives before (shown by the black line) & after (shown by the red line) of C.S immersion in a 1.0 M HCl solution.



### 3.6. DFT and MD simulations

**3.6.1. DFT study.** DFT calculations were used to corroborate the experimental results concerning the inhibition efficiency of the **LF<sub>1</sub>** and **LF<sub>2</sub>** molecules. As the experiments were carried out in a 1.0 M HCl aqueous medium, the first step in the calculation was to identify the protonation site likely to interact with the acid proton. Marvin Sketch<sup>51</sup> software was used for this, with a search carried out over a pH range of 0.0 to 14.0. The results of the protonation analysis are shown in Fig. 10. For all the inhibitors examined, the  $sp^2$  N atom of the benzimidazoles ring is found to be the protonated form, and the S2 structure emerges as the most likely site in acidic medium, with a rate of 100% at pH = 0.0. The S2 form of the inhibitors studied is therefore taken into account in this analysis. The protonated configurations of the inhibitor molecules **LF<sub>1</sub>** and **LF<sub>2</sub>** were generated using the Marvin Sketch program.<sup>52</sup>

Fig. 11 displays the optimized structures, HOMOs, LUMOs, energy gap diagrams, and molecular electrostatic potential maps (ESP) for the inhibitor compounds that are being studied. Upon inspection, it is evident that the HOMO distribution primarily occurs over the benzimidazoles moiety of the inhibitors, whereas the LUMO distribution is limited to the electron-

deficient regions of the benzimidazoles moiety. This indicates that these centers are primarily involved in electron donation and acceptance during the metal–inhibitor interactions.

Table 7 provides a summary of the global reactivity properties of the inhibitor that was studied. Higher  $E_{HOMO}$  values indicate a higher tendency to donate electrons to electron-deficient sites, while lower values indicate a greater ability to accept electrons by the inhibitor.<sup>61,62</sup> As shown, the  $E_{HOMO}$  increases in the order of **LF<sub>2</sub>** > **LF<sub>1</sub>**, indicating that the **LF<sub>2</sub>** inhibitor is more likely to contribute electrons to the virtual 3d orbital of the steel. The  $E_{LUMO}$  decreases in the order of **LF<sub>1</sub>** < **LF<sub>2</sub>**, which means that the **LF<sub>1</sub>** inhibitor molecule has the highest capacity to receive electrons from the steel surface. The smaller the energy gap, the higher the adsorption efficiency. The results show that the  $E_{gap}$  trend is as follows: **LF<sub>2</sub>** (4.801 eV) < **LF<sub>1</sub>** (5.148 eV), indicating that the **LF<sub>2</sub>** inhibitor is the most reactive species and is better able to adhere to the steel surface compared to the **LF<sub>1</sub>** inhibitor. Similar trends can also be observed for the other global chemical reactivity parameters (see Table 7).

Another important factor is the  $\Delta N_{110}$ , showing the transfer of electrons throughout the adsorption process, either from the

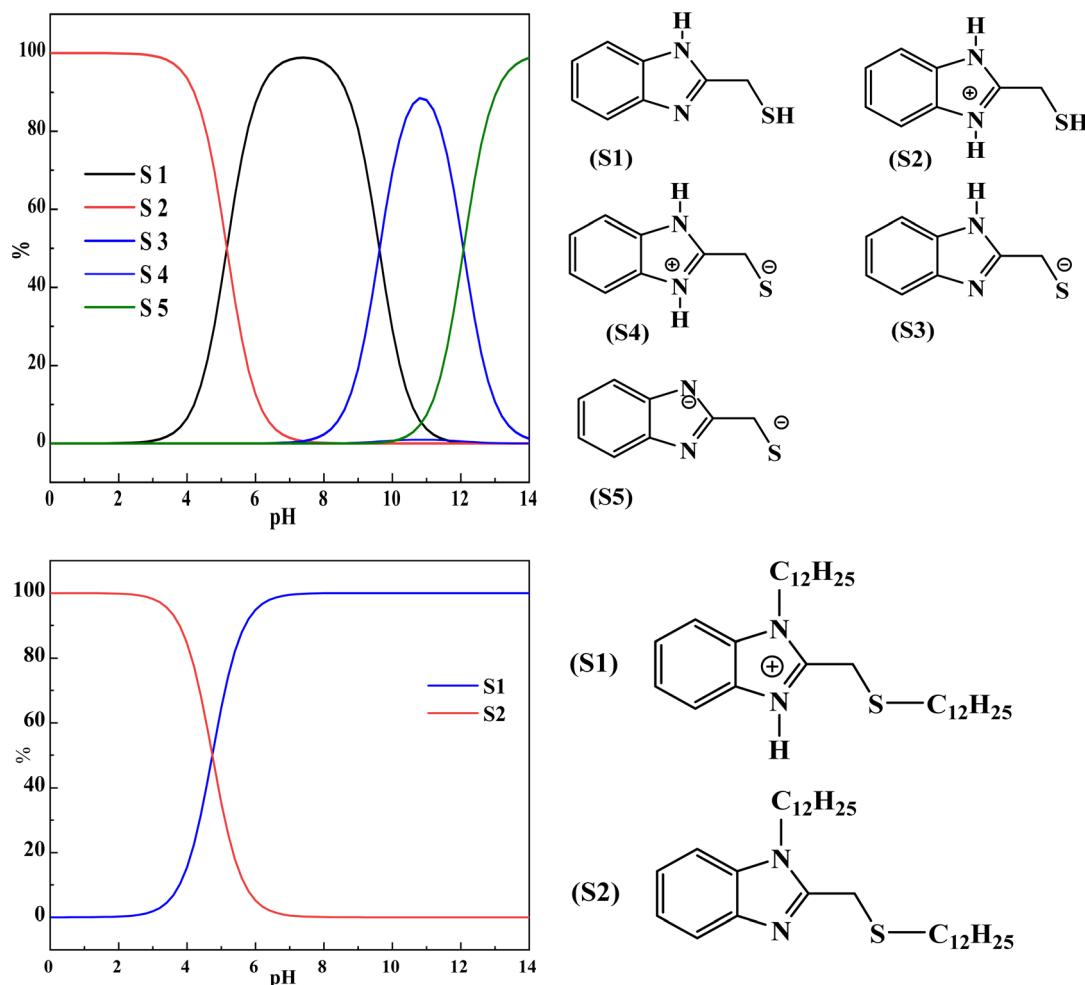


Fig. 10 The protonated configurations of the inhibitor molecules **LF<sub>1</sub>** and **LF<sub>2</sub>**.



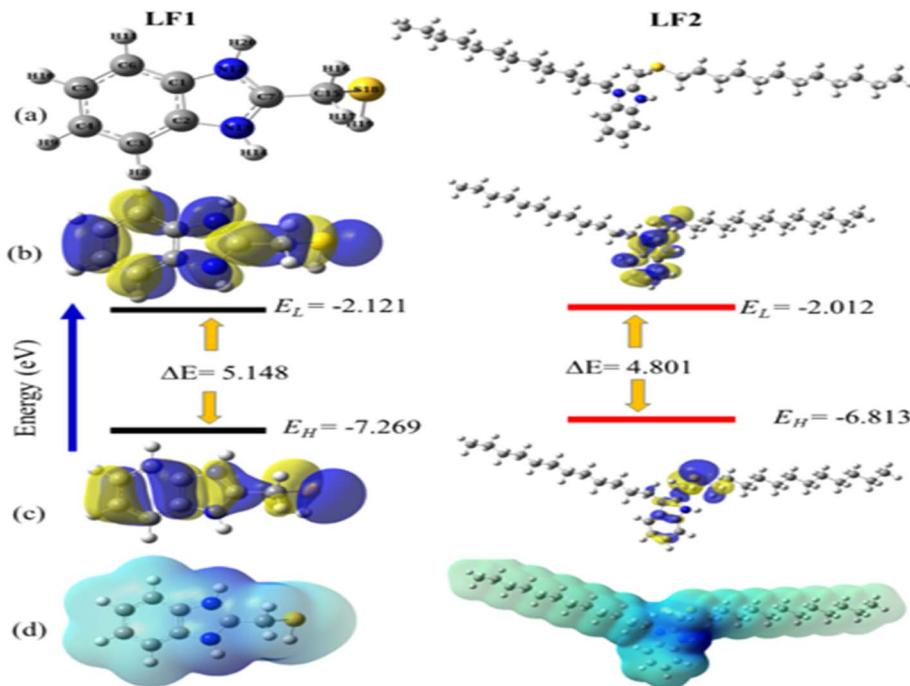


Fig. 11 The optimized geometrical structures (a), LUMO's (b), HOMO's (c), ESP maps (d) & the energy gaps of the investigated species (LF<sub>1</sub> and LF<sub>2</sub>).

Table 7 The computed quantum chemical parameters for LF<sub>1</sub> & LF<sub>2</sub>

	LF <sub>1</sub>	LF <sub>2</sub>
$E_0$ (Ha)	-817.90697	-1761.52369
$E_0^-$ (Ha)	-817.99001	-1761.60258
$E_0^+$ (Ha)	-817.64291	-1761.27386
$E_H$ (eV)	-7.269	-6.813
$E_L$ (eV)	-2.121	-2.012
$\Delta E$ (eV)	5.148	4.801
$\Delta E_1$ (eV)	5.781	5.890
$\Delta E_2$ (eV)	7.118	6.662
IP <sub>v</sub> (eV)	7.185	6.798
EA <sub>v</sub> (eV)	2.260	2.147
$F_g$ (eV)	4.926	4.652
$\chi$ (eV)	4.723	4.472
$\eta$ (eV)	2.463	2.326
$S$ (eV)	0.406	0.430
$\omega$ (eV)	4.528	4.300
$\Delta E_{\text{steel/inh}}$ (eV)	-1.553	-1.398
$\Delta N_{110}$	0.020	0.075
$\Delta E_{\text{b-d}}$ (eV)	-0.616	-0.581

inhibitor (positive values) to the metal surface or the other way around (negative values) when the metal surface or inhibitors are brought into close proximity from a low/electronegativity system to a high-electronegativity system until all chemical potentials are equal. If  $\Delta N < 3.6$ , the summarized results in Table 7 show that the values of the ( $\Delta N_{110}$ ) are positive and  $<3.6$ , illustrating that the inhibition efficiency of the investigated inhibitors is improved, and they have their ability to donate electrons to the metal surface is increased as in Cherinka *et al.* work.<sup>63</sup> Our results show that the trend of the investigated

inhibitors is as follows: LF<sub>2</sub> (0.0747) > LF<sub>1</sub> (0.020). These results show the importance of lengthening the carbon chain in increasing  $\Delta N_{110}$  and, consequently, increasing the fraction of electron transfer from the inhibitor molecule to the metal surface.

The interaction affecting the way inhibitors and the metal surface interact can be treated in view of the energy gaps between the metal and inhibitors as calculated in the following expressions:<sup>61,62,64</sup>

$$\begin{aligned} \Delta E_1 &= E_L^{\text{inh}} - E_H^{\text{Fe}} \\ \Delta E_2 &= E_L^{\text{Fe}} - E_H^{\text{inh}} \end{aligned} \quad (13)$$

where  $E_H^{\text{inh}}$ ,  $E_L^{\text{inh}}$ ,  $E_H^{\text{Fe}}$  and  $E_L^{\text{Fe}}$  are the energies of HOMO & LUMO of the molecule and Fe. In the above equation,  $E_H^{\text{Fe}}$  and  $E_L^{\text{Fe}}$  equal to -7.9024 and -0.151 eV, respectively.<sup>65</sup> The  $\Delta E_1$  term shows the ability of electrons to flow from carbon steel to LUMO of the inhibitor molecule. Whereas  $\Delta E_2$  corresponds to the electrons flow from the inhibitor molecule to carbon steel. As can be noticed in Table 7,  $\Delta E_1 > \Delta E_2$ , indicates that the flowing of the electrons from the carbon steel (Lewis's base) to the vacant orbitals of the inhibitor (Lewis's acid) is energetically favored. Among the inhibitors analyzed the energy gaps  $\Delta E_1$  are higher than  $\Delta E_2$ , aligning with the findings from the experimental phase. These results are consistent with those derived from eqn (12). They demonstrate negative values for  $\Delta E_{\text{steel/inh}}$ , indicating the interaction capability between the metal-surface and inhibitor molecules through back-donation interactions.

**3.6.2. Local reactivity (LR).** Based on the electrical density changes during a reaction employing the natural charges, Fukui functions were employed to determine the most reactive



location for nucleophilic and electrophilic attacks in the inhibitors under probes.<sup>65–70</sup> In order to produce the wave functions, single-point calculations of the  $N, N+1, N-1$ , and  $N-1$  electron systems on the optimized geometries of neutral species were carried out at the same theoretical level. The 3D-surfaces of the Fukui functions and their condensed values based on Hershfield charges were then determined using the Multiwfn<sup>71</sup> program using the following equations:

$$f_k^+ = q(N+1) - q(N) \text{ (for nucleophilic attacks)} \quad (14)$$

$$f_k^- = q(N) - q(N-1) \text{ (for electrophilic attacks)} \quad (15)$$

where the Hershfield charges of the  $N, N+1$ , and  $N-1$  systems are, respectively,  $q(N)$ ,  $q(N+1)$ , and  $q(N-1)$ . The following equation provides the dual Fukui descriptor or second-order Fukui functions ( $\Delta f_k$ ):<sup>72–74</sup>

$$\Delta f_k = f_k^+ - f_k^- \quad (16)$$

The inhibitors under investigation contain several active centers, which is advantageous based on their chemical structures. The inhibitor centers' ability to provide electrons to the steel surface rises with the amount of partially negative active centers.<sup>75</sup> The three-dimensional isosurfaces of the Fukui functions ( $f_k^-, f_k^+$ , and  $\Delta f_k$ ) are shown in the ESI (Fig. S3†). Like the surfaces for HOMOs and LUMOs (Fig. 11), these isosurfaces confirm the findings about the active centers of the species under study. A center is generally thought to be vulnerable to nucleophilic attack if its  $\Delta f_k$  is greater than zero and if  $f_k^+$  is the highest. On the other hand, the situation is the opposite for

electrophilic attack centers, where  $\Delta f_k$  is negative and  $f_k^-$  is the maximum.<sup>76</sup> The main conclusions of the  $f_k^+$ ,  $f_k^-$  and  $\Delta f_k$  condensed Fukui indices for the inhibitors under study are presented in Table 8. A full dataset can be seen in ESI Tables S1 and S2.†

Table 8 makes clear that for protonated species, S18 & C7 are the highly reactive centers that are more suited for nucleophilic assaults, whereas S18 & C4 are more vulnerable to electrophilic ones. Precisely, for the protonated form of **LF<sub>1</sub>**, the centers that may be favored for electrophilic attacks and nucleophilic attacks are S18 & C7, respectively. For the protonated form of **LF<sub>2</sub>**, the centers that may be favored for electrophilic attacks are S18 & C19. Whereas, for the same inhibitor, the sites that may be favored for the nucleophilic attacks are C7 & C3.

**3.6.3. MD simulation.** MD simulation is a method more similar to the experiment, giving more explanatory results for the difference in inhibitory efficacy values obtained by the experiment for the **LF<sub>1</sub>** and **LF<sub>2</sub>** inhibitors.<sup>77</sup> In this case, Fig. 12 depicts the adsorption configurations that are the most stable of all the systems. As shown in Fig. 12, for the protonated forms, the **LF<sub>1</sub>** molecule **LF<sub>1</sub>** (neutral and protonated) adsorbs completely to the Fe(110) surface, while the **LF<sub>2</sub>** molecule **LF<sub>1</sub>** (neutral & protonated) adsorbs only *via* the benzimidazole base molecule, demonstrating that this structure is more effective and reactive, contributing to the effectiveness of inhibition. Furthermore, the alkyl chains in the **LF<sub>2</sub>** molecule are of no importance in terms of chemical reactivity. It should be noted that protonation has no impact on the mode of adsorption, giving the illusion that the two forms react in the same way in the acid medium.

The degree of interaction is well represented by energy parameters such as interaction and binding energies, with a more negative value of  $E_{\text{interaction}}$  indicating a better interaction, while a more positive  $E_{\text{binding}}$  value means better adsorption. These both descriptors are computed based on the following two formulas:<sup>78,79</sup>

$$E_{\text{interaction}} = E_{\text{total}} - E_{\text{inhibitor}} - (E_{\text{surface+solution}}) \quad (17)$$

$$E_{\text{binding}} = -E_{\text{interaction}} \quad (18)$$

In Table 9, a greater negative value for  $E_{\text{interaction}}$  indicates a higher level of protective adsorption due to increased interaction between the inhibitor and steel surface.<sup>80</sup> As noted in Table 9, the Fe(110)/**LF<sub>2</sub>** (neutral) system bears a more negative value for  $E_{\text{interaction}}$  ( $-1184.209 \text{ kJ mol}^{-1}$ ) and a more positive value for  $E_{\text{binding}}$  ( $1184.209 \text{ kJ mol}^{-1}$ ), showing that the neutral molecule **LF<sub>2</sub>** interacts more strongly with the first Fe(110) layer. This result and all the data contained in Table 9 confirm the results generated experimentally.

This method's primary goal is to use the radial distribution function (RDF) to assess the nature of the bonding and adsorption with the neutral and protonated atoms in the inhibitor's initial iron layer. Estimating the interatomic distance between the atoms and heteroatoms of the forms under study and Fe(110) is a particularly important use of this method.<sup>81</sup> Prior studies have verified that chemical adsorption

**Table 8** The most important condensed Fukui indices for the **LF<sub>1</sub>** & **LF<sub>2</sub>** inhibitors. The table is sorted according to the most reactive centers

Center	$f_k^+$	Center	$f_k^-$	Center	$\Delta f_k$
<b>LF<sub>1</sub> neutral</b>					
S18	0.1401	C4	0.1306	C4	-0.0586
C7	0.0898	N12	0.091	C1	-0.0531
C3	0.0853	C1	0.0886	C2	-0.0325
C6	0.0825	C6	0.0868	S18	0.0921
N13	0.072	C7	0.0825	C15	0.0173
N12	0.0657	C3	0.0759	C3	0.0093
<b>LF<sub>1</sub>-protonated</b>					
S18	0.1451	S18	0.331	S18	-0.1859
C7	0.127	C4	0.08	C1	-0.037
C3	0.0683	C1	0.0621	C2	-0.0267
C6	0.0679	C5	0.0593	C7	0.0894
N13	0.0642	C2	0.0519	N13	0.0406
N12	0.0634	C7	0.0375	C3	0.0343
<b>LF<sub>2</sub>-protonated</b>					
C7	0.1178	S18	0.4558	S18	-0.3646
S18	0.0912	C19	0.0375	C19	-0.0308
C3	0.0759	C15	0.0329	C15	-0.0066
C6	0.0722	C4	0.0304	C7	0.1113
C5	0.0612	C5	0.0212	C3	0.0601
N13	0.0584	C1	0.0206	C6	0.0564



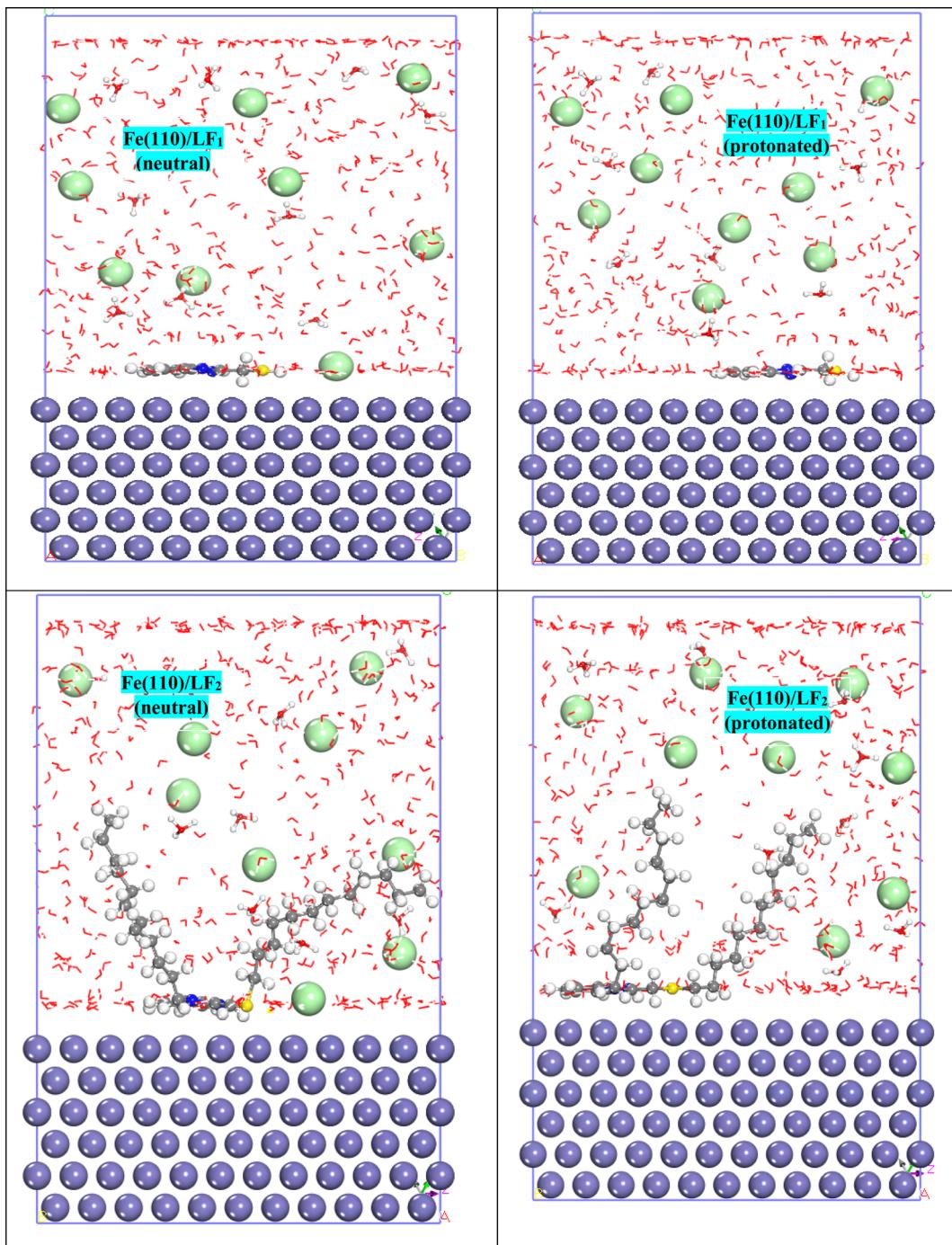


Fig. 12 Adsorption configurations feature the most stable forms for the systems  $\text{Fe}(110)/\text{LF}_1$  (neutral),  $\text{Fe}(110)/\text{LF}_2$  (neutral),  $\text{Fe}(110)/\text{LF}_1$  (protonated), and  $\text{Fe}(110)/\text{LF}_2$  (protonated).

Table 9 Calculated interaction energy of each simulated system (all in  $\text{kJ mol}^{-1}$ )

Systems	$\text{Fe}(110)/\text{LF}_1$ (neutral)	$\text{Fe}(110)/\text{LF}_2$ (neutral)	$\text{Fe}(110)/\text{LF}_1$ (protonated)	$\text{Fe}(110)/\text{LF}_2$ (protonated)
$E_{\text{interaction}}$	-1138.360	-1184.209	-1144.982	-1180.681
$E_{\text{binding}}$	1138.360	1184.209	1144.982	1180.681

is more probable for bond lengths shorter than  $3.5 \text{ \AA}$ , whereas physical adsorption is suggested by longer link lengths. All bond lengths have values smaller than  $3.5 \text{ \AA}$ , as seen by the initial peaks in Fig. 13, indicating chemical adsorption.

### 3.7. Corrosion inhibition mechanism

The protective mechanism of these compounds on the C.S can be explained through the inhibition process exerted by the

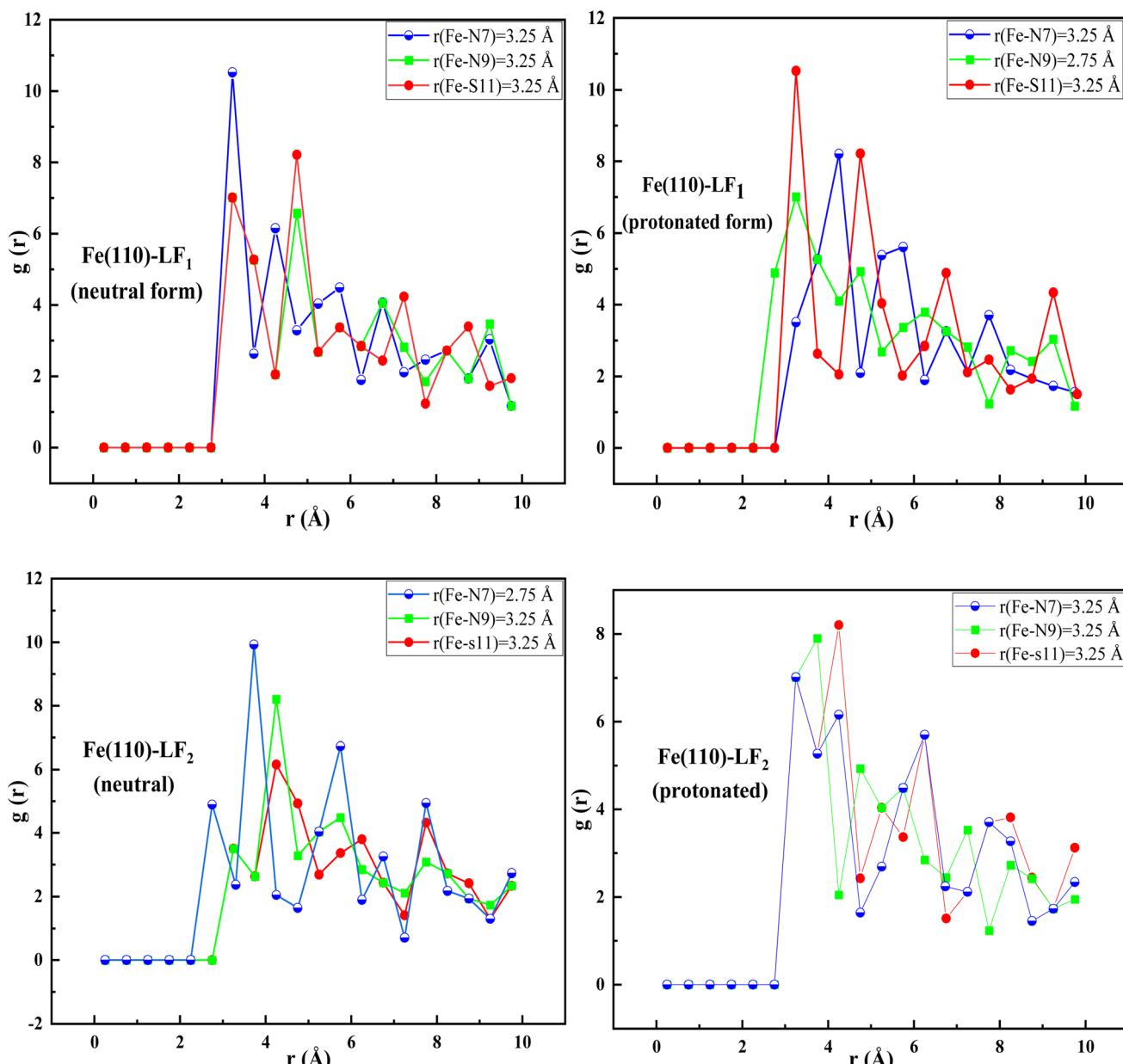


Fig. 13 RDF of inhibitors neutral and protonated onto Fe(110).

inhibitor over the metal surface. The main modes of adsorption of an organic molecule onto a metal surface include chemisorption, physisorption, or a combination of the two. Many parameters, such as the concentration of the inhibitor, the presence of active sites, the charge of the molecules, and the characteristics of the metal surface, influence the adsorption processes of the inhibitor.

The benzimidazole compounds studied in this research have functional groups containing S, N heteroatoms and phenyl rings, which can act as adsorption sites for these inhibitors. Fig. 14 shows a potential inhibition process to provide a more detailed explanation of the corrosion inhibition mechanism of C.S in 1.0 M HCl solution.

Previous studies have established that C.S develops a positive surface charge when exposed to an acidic HCl solution. The

$\text{Cl}^-$  ions present in the solution promote the formation of a negatively charged surface on the initially positively charged C.S, facilitating the adsorption of cations from the solution. Due to the presence of an unshared pair of electrons on the nitrogen atoms, benzimidazole molecules can be protonated in an acidic solution. Following an electrostatic interaction, or physisorption, the protonated molecules ( $\text{LF}_1\text{H}^+$  and  $\text{LF}_2\text{H}^+$ ) can be adsorbed onto the surface of the C.S, as shown in Fig. 14.

However, the presence of two alkyl chains ( $\text{C}_{12}\text{H}_{25}$ ) in  $\text{LF}_2$  provides a larger surface area for adsorption onto the steel surface, this increased surface area facilitates stronger interactions between the  $\text{LF}_2$  molecules and the steel surface, leading to improved inhibition efficiency. Also, the longer alkyl chains in  $\text{LF}_2$  often contribute to increased hydrophobicity of the  $\text{LF}_2$  molecules, which enhanced hydrophobic interactions between

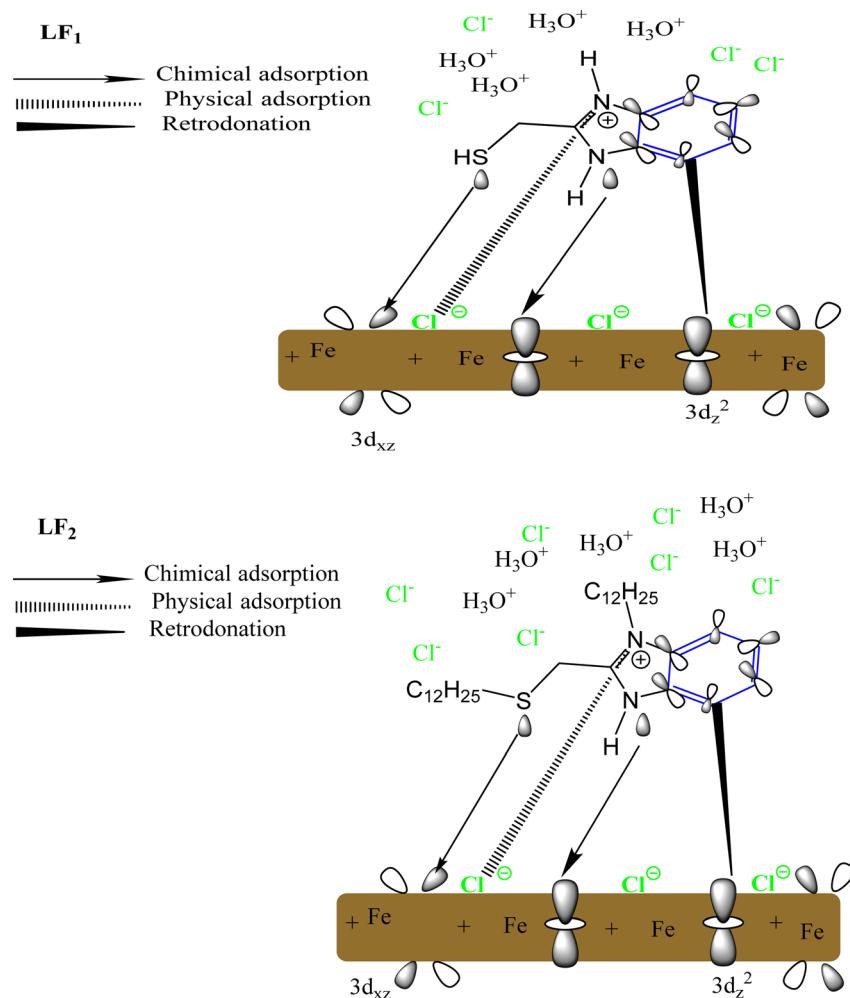


Fig. 14 Proposed mechanism of corrosion inhibition of  $\text{LF}_1$  and  $\text{LF}_2$ .

the  $\text{LF}_2$  molecules, and the steel surface can improve the stability of the protective film formed and reduce the penetration of  $\text{Cl}^-$ .<sup>82</sup>

The increased negative charge on the C.S surface is transferred from the vacant  $\pi^*$  orbital of the inhibitor molecules to the d-orbital of Fe, leading to stronger inhibitor adsorption on the steel surface. Additionally, there is a possibility that the d-orbitals on the C.S surface can donate electrons back to the aromatic cycles. Consequently, the steel surface eventually forms an adsorbed layer of the inhibitor, which acts as a protective barrier between the metal and the corrosive medium, preventing metal corrosion.<sup>83</sup>

## 4. Conclusion

The effectiveness of two new benzimidazole compounds as corrosion inhibitors for C.S in a 1.0 M HCl environment was evaluated through computational and experimental studies. The following are the main conclusions:

- The results showed that in a 1.0 M HCl solution,  $\text{LF}_1$  and  $\text{LF}_2$  efficiently serve as corrosion inhibitors for carbon steel substrates. With increasing inhibitor concentration, both chemicals' inhibition efficiency rise in the following order:  $\text{LF}_1 < \text{LF}_2$ .

- The results obtained from potentiodynamic polarization data suggest that the investigated compounds are mixed type inhibitors.

- The electrochemical impedance spectroscopy plots revealed the adsorption of both imidazole molecules, which is confirmed by the rise of the polarization resistance and the reduction of the values of the double layer capacitance.

- SEM/EDX/AFM/contact angle analysis confirms the creation of a protective film on the C.S-surface.

- The research in the UV-visible clearly showed the creation of complexes.

- The acquired result through DFT/MDs simulation strongly corroborated the findings from the laboratory experiments.

## Data availability

The data supporting this article have been included within the manuscript and its ESI.†

## Conflicts of interest

There are no conflicts to declare.



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