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A "turn-off" photoluminescent sensor for H₂O₂ detection based on a zinc oxide–graphene quantum dot (ZnO–GQD) nanocomposite and the role of amine in the development of GQD†

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In this work, graphene quantum dots (GQD) were prepared through a hydrothermal process. The photoluminescence (PL) emission spectrum for GQD prepared with high NH₄OH concentration (sample D1-t) was attained at lower wavelength (406 nm), compared to GQD synthesized with low NH₄OH concentration (sample D2-t attained at 418 nm). From these results, a smaller particle size for D1-t was deduced; according to TEM images the GQD particles are around 5 nm. The Raman I_{D3}/I_G ratio which is related to C–O groups at the edges of GQD and the full width at half maximum was lower for D1-t than D2-t. This was ascribed to the amine group incorporation at the edges and at the basal planes in D1-t, whilst in D2-t they prefer principally the edges of the GQD structure. The ZnO nanoparticles bonded to GQD (ZnO–GQD, nanocomposites) enhance the PL emission intensity. The H₂O₂ detection tested by photoluminescence spectroscopy, was found to occur thanks to the ZnO from the nanocomposite and its interaction with H₂O₂, producing a quenching effect. This quenching was accentuated by the increase of the H₂O₂ concentration. Such properties suggest the ZnO–GQD nanocomposite as a candidate to be used as a sensor material.

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1 Introduction

Graphene oxide is obtained by the oxidation of graphene, and this process is essential for this material to be used in transistors, photodetectors, electronic displays, among others.¹ The particle size reduction of graphene oxide leads to obtain what are known as graphene quantum dots (GQD). These GQD have lateral dimensions from 3 nm up to 100 nm having attributes of edge effects, quantum and electron confinements.^{2,3} These characteristics allow the GQD to be potentially used as materials for fluorescence imaging of organic tissues,⁴ drug delivery⁵ and sensors.^{6,7} The hydrothermal (HT) process is a recurring method to prepare GQD. The HT method has been used with water, chemical bases and acids, but one of the most effective solvents

is the H₂O₂ which decomposes in 'OH and 'O[−] radicals, producing the cut and the development of epoxy and carbonyl groups, respectively.⁸ This method also induces the functional groups to link at the edges of GQD which serve as anchorage sites for polymers or metal oxides.⁹ In particular, when nitrogen functional groups are aggregated in the GQD structure, the photoluminescent properties are enhanced (*i.e.* increment in emission intensity).¹⁰

Moreover, the incorporation of metallic particles to the graphene oxide conduces to the enhancement when its sensor properties are evaluated.¹¹ Several research works for sensor applications have examined GQD along with metals as gold and platinum,¹² silver¹³ and titanium oxide.¹⁴ Particularly, these composites (metals and metals oxides–GQD) have been used in H₂O₂ detection. The H₂O₂ is a molecule related to the degradation of some physiological functions which conduces to cardiovascular diseases, Alzheimer, atherosclerosis, cellular oxidative stress, cellular ageing and cancer.^{15–17} For this reason, the early and precise H₂O₂ detection is essential for an opportune diagnosis.

Numerous sensors have been developed for the H₂O₂ detection and they are mainly based in analytic methods as electrochemical (enzymatic and non-enzymatic), high resolution chromatography, chromatography mass spectroscopy, colorimetry, capillary electrophoresis, infrared absorption

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spectroscopy and photoluminescence spectroscopy.^{15,18–20} The photoluminescent sensors offer a myriad of advantages as operative simplicity, high sensibility, high selectivity and fast response.^{21,22}

Recently nanomaterials (nanoclusters, nanorods, nanospheres) of noble metals as gold and silver, metal–ligand complexes, organic colourants, metals oxides nanoparticles and *quantum dots* of CdSe, carbon, ZnS and graphene, have been assessed as photoluminescent sensors.^{21–25} There are also some reports over the H₂O₂ detection by means of photoluminescence using GQD. For instance, He *et al.* developed a photoluminescent biosensor of GQD functionalized with *hemine* to monitor H₂O₂ and glucose in human serum.²⁶ A TiO₂–GQD nanocomposite was used for H₂O₂ detection by photoluminescence spectroscopy, using an excitation wavelength of 360 nm having well detection in a range from 10^{–14} M to 10^{–2} M.¹⁴

Photoluminescence studies for the detection of H₂O₂ still are scarce and to the best of our knowledge there are not works concerning ZnO–GQD nanocomposites for this purpose. This study contributes with a general knowledge from the synthesis of GQD with amine groups and the ZnO–GQD nanocomposite which is important since some of its advantages are the simplicity of the synthesis and the economic feasibility compared to noble metals. Therefore, in this study, it is presented an analysis over the role of NH₄OH during the synthesis of GQD and ZnO–GQD and their performance in the H₂O₂ detection.

2 Experimental

2.1 Chemical substances and materials

In this investigation the following substances and materials were used. From Sigma-Aldrich: synthetic graphite with particle size < 20 µm, 30% H₂O₂ and dialysis tubing benzoylated, 32 mm width. From Fermont (Monterrey Mexico): 99.7% KMnO₄, 28% NH₄OH, 99.3% Zn(CH₃COO)₂·2H₂O. Concentrated H₂SO₄ (DEQ, Monterrey, Mexico). 85% H₃PO₄ (CTR, Monterrey, Mexico). Deionized water (Karyeth reactivos, Monterrey, Mexico) used was 0.76 µS cm^{–1}.

2.2 Graphene oxide synthesis

Graphene oxide was synthesized by the modified method of Hummers similar to Yadav *et al.*²⁷ and it is briefly described below. In a beaker set on an ice bath, 0.4 g of graphite, a volume of 53 mL of concentrated H₂SO₄, and 13.5 mL of H₃PO₄ were placed and gently stirred. After one hour, 2.9 g of KMnO₄ were carefully added maintaining the stirring during 15 min. The beaker was removed from the ice bath and taken to heat at 50 °C keeping a moderately stirring for 10 h. Upon completion, the mixture was prudently decanted and the residual solid was filled with 300 mL of deionized water and 4 mL of H₂O₂. After 15 min of stirring this mixture was taken to centrifugation at 6000 rpm for 30 min. The solid obtained was washed with deionized water until the rinsing water pH was 6. The solid then was placed in a dialysis tubing to eliminate the excess of ions.

Finally, this sample was taken to centrifugation, decanted and the solid was dried in an oven at 40 °C during 5 h to later being stored in a glass container. The final sample was identified as GOx.

2.3 Synthesis of graphene quantum dots

For the sake of simplicity the Table 1 summarizes the synthesis routes followed in this section, still the detailed synthesis is described below. Graphene quantum dots (GQD) were prepared by hydrothermal process and 3 routes were used. In all the experiments deionized water and a GOx mass of 0.0300 g were used for each route.

2.3.1 Route 1. The solid and 8 mL of a mixture of NH₄OH–water (with 1 : 3 volume ratio) were added to a beaker. Later, the content was transferred into a 50 mL of Teflon autoclave and this was introduced to an oven at 170 °C over 6 h. Once the autoclave was at room temperature the product was filtered and the liquid was named D1-w and stored, whilst the solid was kept for ulterior characterization. A volume ratio of 1 : 1 from D1-w and water, was heated at 55 °C during 40 minutes to remove the excess of NH₄OH and the product obtained was identified as D1-t. In this way, the samples without heat treatment are identified by “w” suffix and those treated with “t” suffix.

2.3.2 Route 2. In a beaker the specified mass of GOx was mixed with 15 mL of a NH₄OH–water mixture (with a 1 : 10 volume ratio). This mixture was transferred to a 50 mL Teflon autoclave and introduced into an oven at 170 °C during 6 h. When the autoclave reached room temperature the product was filtered and the liquid, named D2-w, was stored. A heat treatment, as described in route 1, was also carried out on this sample to finally obtain the D2-t.

2.3.3 Route 3. To verify the effect of an oxidant in GOx, the solid was blended with 15 mL of a H₂O₂–water mixture (1 : 5 volume ratio). This mixture was introduced into the Teflon autoclave and placed into an oven at 170 °C for 6 h. The reactor cooled down and the solid was found to be completely dissolved. The obtained product was identified as H-GOx.

Subsequently, 8 mL of a NH₄OH/H-GOx mixture (1 : 12 volume ratio) was introduced into Teflon autoclave and placed in an oven at 170 °C for 6 h. After completion, a small fraction of black solid was emerged. The liquid was kept and named as D3-w, which was not taken to posterior treatment.

2.4 Synthesis of ZnO nanoparticles

For the synthesis of ZnO nanoparticles (ZnNP), 50 mL of a 0.125 M solution of Zn(CH₃COO)₂·2H₂O was placed into a beaker magnetically and vigorously stirred. The initial pH of this solution was 7. A NaOH solution (0.5 M) was added (drop by drop) to the zinc acetate solution. Once the pH reached 12, the solid had a bright white jelly aspect, and a strong stirring was maintained for 1 h. The solid was washed with deionized water and a centrifugation process was taken until a pH of 7 was measured from rinse water. The humid solid was dried at 80 °C for 20 min. Finally, this solid was identified as ZnNP and stored for future tests and to prepare the composite with sample D1.



Table 1 Identification of the GQD prepared by the routes described in section 2.3

Route	GQD precursor	Solvent used	Solvent concentration	Samples obtained	
				Without treatment	After treatment
1	GOx	NH ₄ OH	High	D1-w	D1-t
2	GOx	NH ₄ OH	Low	D2-w	D2-t
3	GOx	H ₂ O ₂	Low	H-GOx	—
3	H-GOx	NH ₄ OH	Low	D3-w	—

2.5 Synthesis of ZD1-w and ZD1-t nanocomposites

A suspension of approximately 0.5% w/v of ZnNP in water was placed in an ultrasonic bath for 30 min. From D1-w and D1-t, 3 mL were diluted in 10 mL of deionized water. A volume of 2 mL from the ZnNP mixture was transferred into each diluted samples and the final mixture was stirred for 25 min. The prepared nanocomposites were named as ZD1-w and ZD1-t.

2.6 X-ray diffraction

X-ray diffraction (XRD) patterns for graphite and GOx were carried out in a Bruker D2 Phaser equipment, with a step size of 0.1° s^{-1} whilst for ZnO a Siemens D5000 equipment was used with a step size of $0.05^\circ \text{ s}^{-1}$. Both equipments employed a copper source with $\lambda = 1.5418 \text{ \AA}$.

2.7 Fourier transform infrared-ATR

In order to verify the functional groups from GOx and D1 sample, the attenuated total reflectance (ATR) adapted to a Fourier transform infrared (FTIR) equipment, was measured. A PerkinElmer Spectrum One spectrometer was used for this objective.

2.8 UV-Vis

Electronic transitions from several samples were identified by UV-Vis using a Shimadzu UV-1800 equipment.

2.9 Photoluminescence spectra

Photoluminescence (PL) spectra were recorded in a PerkinElmer LS55 spectrophotometer. The following modalities were required.

2.9.1 Comparison among routes of synthesis. The PL spectra were taken in aqueous phase for all samples prepared in section 2.3 without any dilution, except for the route 1. For the route 1, 3 mL of the raw sample were diluted in a 25 mL volumetric flask. For all the samples, the emission spectra were measured varying the excitation energy every 25 nm from 250 to 375 nm and they are shown in Fig. 4.

2.9.2 Emission spectra for H₂O₂ detection. These spectra were taken for the samples D1-w and D1-t and their parents ZD1-w and ZD1-t. In this regard, H₂O₂ solutions were prepared by varying concentration from 10^{-2} M to 10^{-14} M . From each H₂O₂ solution a volume of 1 mL was added to a beaker; then 1 mL of the corresponding sample and 1 mL of deionized water

were incorporated. After 5 min of mild stirring, the emission spectra were measured and these are exhibited in Fig. 5.

2.10 Raman spectroscopy

Raman spectra were taken on a dispersive Raman instrument Senterra from Bruker, equipped with a 785 nm laser and an Olympus microscope. A $20\times$ objective was employed; the integration times varied from 2 to 8 s depending on the band intensity, and the number of co-added spectra was between 8 and 10. The spectral range of detection went from 70 to 3500 cm^{-1} with a spectral resolution of $9\text{--}15 \text{ cm}^{-1}$.

2.11 SEM

With the aim to highlight the modifications between graphite (GF), graphene oxide (GOx) and graphene quantum dots (GQD) from sample D2-w, a scanning electronic microscopy (SEM) analysis was achieved. For this purpose, a JEOL JSM 6701F equipment was used. The images are shown in Fig. S3 from the ESI.[†]

2.12 TEM

The sample for transmission electron microscopy (TEM) was prepared by dispersing the solids in distilled water, placed and dried on a TEM grid. Afterwards, they were observed with a JEOL JEM 1010 with 60–80 kV microscope.

3 Results

3.1 Chemical structure study

An important matter of this study was the synthesis effectiveness. Graphene oxide structure pattern was obtained by XRD, compared with graphite and these patterns are presented in Fig. 1.

The diffraction pattern of graphite concur with JCPSD card 04-007-2076 where the peak of this solid was found at $2\theta = 26^\circ$. In accordance with this card, the structure of graphite is rhombohedral which interplanar distance is $d = 0.335 \text{ nm}$.²⁸ After the synthesis of GOx a change in the graphite structure is evidenced (Fig. 1); this pattern, shows a peak around $2\theta = 10^\circ$ which normally is observed when the Hummers' method is used for the synthesis of graphene oxide.^{27,29} In Fig. 1 the diffraction pattern from ZnNP coincides with JCPSD card 01-007-2551.³⁰ The diffraction patterns from GOx and ZnNP provide evidence that the required compounds were obtained.



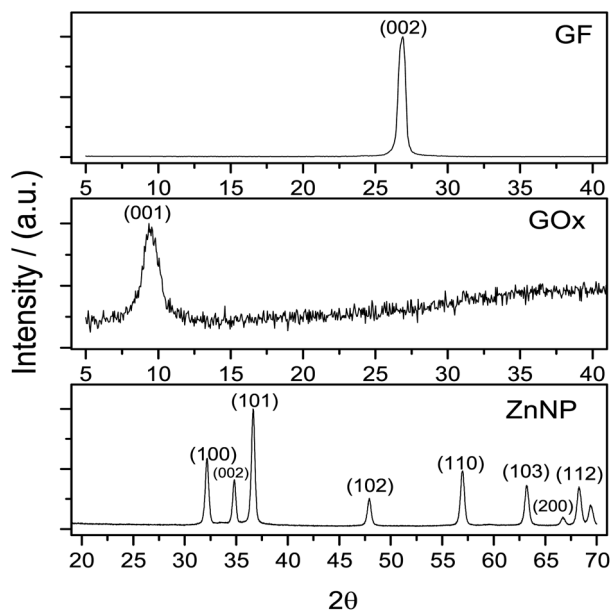


Fig. 1 Normalized XRD diffraction patterns from graphite (GF), graphene oxide (GOx) and zinc nanoparticles (ZnNP). Graphite has a characteristic peak at $2\theta = 26^\circ$ whilst the GOx presents its peak at $2\theta = 10^\circ$. XRD pattern from ZnNP exhibited characteristic peaks at the angles $2\theta = 31, 34, 36, 47, 56, 62, 66, 68$ and 69° .

3.2 Functional groups of GQD and their interaction with ZnNP

3.2.1 Identification of functional groups with FTIR. In order to contrast the bands derived from the incorporation of the nitrogenated ligand to GOx an ATR-FTIR study was carried

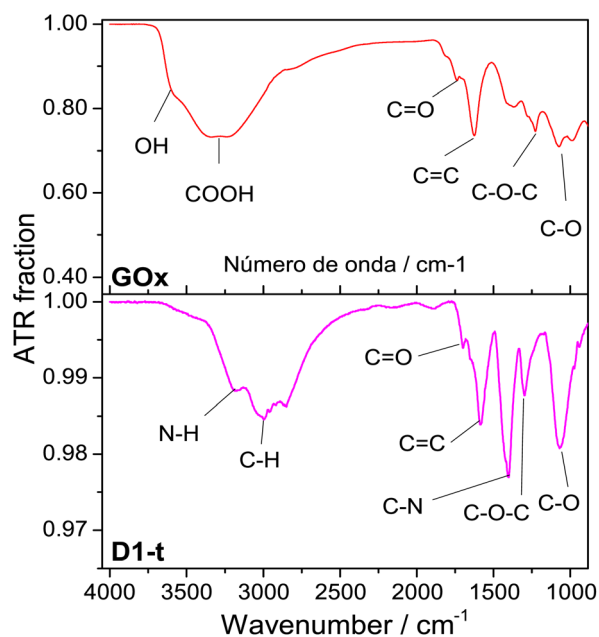


Fig. 2 ATR normalized spectra from GOx and D1-t. Measurements were done at room temperature. Bands assignments in each sample are identified below spectra.

out and the results are exhibited in Fig. 2. Clearly, GOx presents a series of bands attributed to diverse types of bonded species. The band at $\approx 3600 \text{ cm}^{-1}$ is attributed to the stretching of the O-H bond. This O-H might be positioned at the basal plane from graphene oxide, linked to zones where C-C bonds are weak. As well, the C=O band is shown at $\approx 1733 \text{ cm}^{-1}$, the C=C band at $\approx 1629 \text{ cm}^{-1}$, the C-O-C at $\approx 1230 \text{ cm}^{-1}$ and lately the band assigned to C-O is shown at $\approx 1077 \text{ cm}^{-1}$.^{31–33}

From D1-t spectra, two new bands which are associated to the nitrogen incorporation to the GQD structure, are highlighted. One of these is shown at $\approx 1407 \text{ cm}^{-1}$ and is assigned to the C-N bond and other at $\approx 3180 \text{ cm}^{-1}$ from N-H bond. Both are characteristic of the amine groups.³² Note that the band at $\approx 3600 \text{ cm}^{-1}$ attributed to O-H has vanished in the GQD spectra.

3.2.2 UV-Vis spectra on the heat treatments and GQD-Zn interaction. The presence of the amine functional groups influences not only the photoluminescence properties and the H_2O_2 detection, but it also emerges in the UV-Vis bands. Fig. 3 presents the UV-Vis spectra from the D1 samples set, with and without heat treatment, as their derivatives ZD1 composites.

The band centred at 234 nm, corresponds to the electronic transitions from $\pi \rightarrow \pi^*$ orbitals of the C=C bonds which possesses principally sp^2 hybridization.^{34,35} A band centred at 296 nm attributed to $n \rightarrow \pi^*$ transition of the C=O bond, emerges in all samples. This high intensity in D1-w and ZD1-w might be influenced by a greater amount in amine groups;³⁵ whilst in D1-t and ZD1-t, even though the amine group are evidenced in the FTIR, this band has diminished and is not visible, respectively. At 258 nm and 334 nm, two bands are displayed in ZD1-t and very tenuous in ZD1-w which are assigned to ZnNP.³⁶ With the aim to observe the absorbance

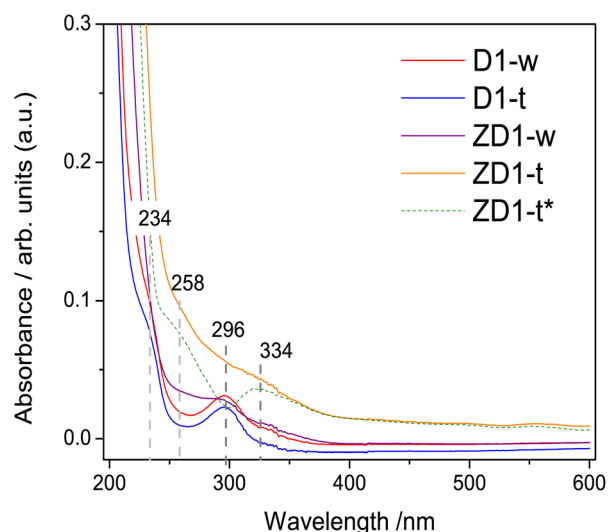


Fig. 3 UV-Vis absorbance spectra from non-treated samples: D1-w and ZD1-w and treated: D1-t and ZD1-t. Spectra were taken in aqueous phase and at room temperature. Data shown over dashed gray lines indicate the corresponding values of bands in nanometers. For ZD1-t* (dashed green line) the shown spectrum is the subtraction of D1-t from ZD1-t.

spectrum from the ZnNP in the aqueous phase, the D1-t spectrum was subtracted from ZD1-t and the bands centred at 258 and 334 nm are more clearly noticed (Fig. 3 ZD1-t*).

3.3 Fluorescence analysis

3.3.1 Comparison of the different routes of synthesis. Photoluminescence emission spectra of GQD, prepared by several routes, are shown in Fig. 4.

Photoluminescence measurements started with an excitation wavelength (λ_{ex}) of 250 nm and its variation, led to gradually increase the emission intensity to a maximum at the $\lambda_{\text{ex}} = 300$ nm for D3-w and at the $\lambda_{\text{ex}} = 325$ nm for the rest of the samples. From $\lambda_{\text{ex}} = 350$ nm the emission intensity begins to decline. Therefore, for any of the routes the emission intensity of GQD, is in function of the excitation wavelength and the variation in the emission is similar for all routes. Another similar aspect in those samples is the peak position with respect to the excitation wavelength parameter; generally, as this parameter increases there is a red-shift. This behaviour is known as the *giant red-edge effect* which is a particular feature of graphene oxide and arises when polar molecules from solvents as water, are not aligned with fluorophores. As these fluorophores are exciting, water molecules align producing an additional step in the energy emission during the relaxation.³⁷ In this regard, some differences might be noted from all the routes, by the analysis of the maxima intensity peaks. In D1-w, D2-w and H-GOx (Fig. 4a–c) the peak of maximum intensity is present at the $\lambda_{\text{ex}} = 325$ nm. This peak shows a blue-shift which

is linked to the higher concentrations of ammonium hydroxide and hydrogen peroxide used during the synthesis. For instance, for D2-w the maximum peak is centred around 418 nm while for D1-w is at 406 nm. This blue-shift might be caused due to a change in the particle size from GQD; this follows a similar trend to that theoretically deduced by Robertson and O'Reilly (1987) whose found that the energy gap between states π and π^* , grows owing to a diminishing in particle size with sp^2 hybridization.³⁸ Therefore, in D1-w the formation of smaller particles is favoured and the same it is devised for H-GOx (Fig. 4).

The purpose of the H-GOx synthesis, was for hydrogen peroxide to split the sheets from graphene oxide reducing its size to a maximum. Comparing H-GOx with D1-w, for $\lambda_{\text{ex}} = 325$ nm, there is a slight blue shift at 392 nm (Fig. 4c). Returning to hypothesis regarding the change in particle size, it is deduced that a reduction in particle size occurred during the H-GOx synthesis. However, the emission intensity from H-GOx considerably decreased, indicating that the introduced functional groups have weak photoluminescent properties; thus, it is inferred that the size does not influence the intensity, as much as the incorporated nitrogenous ligands. This can be corroborated from the subsequent incorporation of amine group, through the reaction with NH_4OH (route 3) in H-GOx. Apparently, the amine group now are linked to different particles that unveil its maxima emission when are excited at 300 nm (Fig. 4d). Therefore for the evolution of GQD, besides the particle size reduction of GOx, the reaction with ammonium hydroxide is necessary to generate this amine group linked to

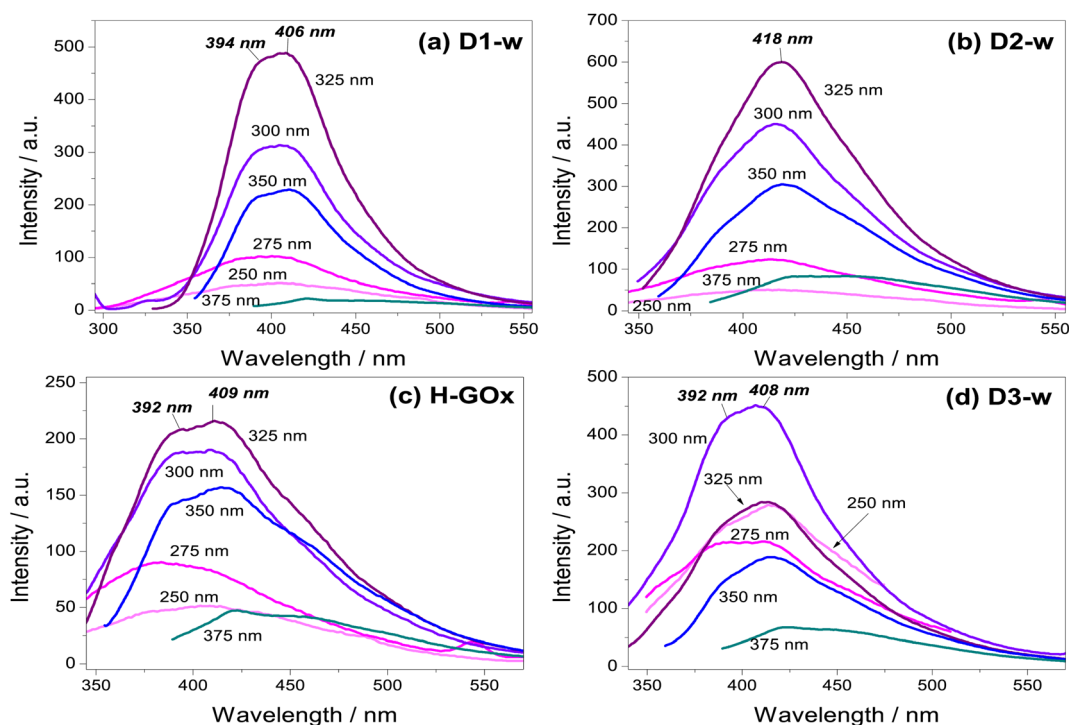


Fig. 4 Emission spectra taken in aqueous phase to compare samples (a) D1-w (b) D2-w, and to observe the effect from (c) H-GOx in (d) D3-w. Values inside each graph for a particular peak of emission are the excitations wavelengths (λ_{ex}). In cursive and bold are shown the peaks position of the maximum emission, only at the $\lambda_{\text{ex}} = 300$ nm for D3-w and at the $\lambda_{\text{ex}} = 325$ nm for the rest.



GQD structure in D3-w. It is worth mentioning that interaction among GQD and the amines ligands, is essential for a maximum emission intensity to exist, due to neither the graphene oxide nor ammonium hydroxide present this characteristic (see Fig. S1 in ESI†).

It has been found that the hydroxyl ions from graphene oxide interact with polar molecules of water (giant red-edge effect), whilst the carboxyls can be protonated with hydrogen ions.³⁹ Because for this $\lambda_{\text{ex}} = 375$ nm the giant red-edge effect is barely noted, it is deduced that in all cases the hydroxyl and carboxyl ions would be scarce. However, since the D1-w has the lowest intensity, it is claimed that in D1-w not only a decrement in particle size has occurred, but a facile incorporation from amine groups resulted.

3.3.2 Fluorescent spectra for the H_2O_2 detection. Now the photoluminescence results from GQD and ZnO-GQD nanocomposites for H_2O_2 detection are presented and for this purpose the GQD from D1 sample were used. The evaluation started by measuring a fresh sample (H_2O_2 free) and then continued by mixing with H_2O_2 solutions from a concentration of 10^{-14} M to 10^{-2} M. The fresh samples (purple lines) indicate in all cases that the elimination of the excess of NH_4OH increases the emission intensity (compare Fig. 5a with 5c and b with 5d). If the presence of zinc is examined, the intensity for the fresh samples also increases in each case; however if the purpose of the detection of H_2O_2 is to find a trend (*i.e.* gradual decreasing or increasing in emission intensities) some differences and “anomalies” are found in PL spectra from Fig. 5.

The fresh samples D1-w and ZD1-w, initially present an emission intensity of 120 and 300 a.u. respectively (Fig. 5a and b). When the H_2O_2 concentration rises, there is not a trend in D1-w and its emission intensity remains almost the same. For the 10^{-8} M concentration, there is a slightly increment in the intensity regarding the previous. In ZD1-w a mild trend to decrease the intensity is exhibited as the H_2O_2 concentration rises. Nevertheless, this occurs until the concentrations of 10^{-6} M and 10^{-10} M are analysed; for these solutions there is also an increase in the emission intensity, reaching almost the 350 a.u. These emission increments are related with the chemical environment surrounding the particles of GQD. An excess of NH_4OH produces a chemical reduction from GQD carboxylates (see $I_{\text{D1}}/I_{\text{G}}$ ratio in Table 2). This reduction, increases the availability of the GQD as individuals fragments with sp^2 domains, which are susceptible of being enveloped by greater particles⁴⁰ (see explanation in section 4.2).

Likewise as D1-w, the sample D1-t does not show a notable trend to diminish the emission intensity as the H_2O_2 increases, however it is not observed any increment in the emission for the subsequent spectra respecting the fresh spectrum (Fig. 5c). The emission intensity in ZD1-t decreases gradually as the H_2O_2 grows, except for the concentrations of 10^{-14} M and 10^{-12} M which not show a normal performance. Notwithstanding, in ZD1-t there is a proper trend to decrease the emission and therefore an adequate H_2O_2 detection. This enhancement is attributed to: (1) thermal treatment which eliminates the presence of ammonia reducing the effect of the unusual increase in

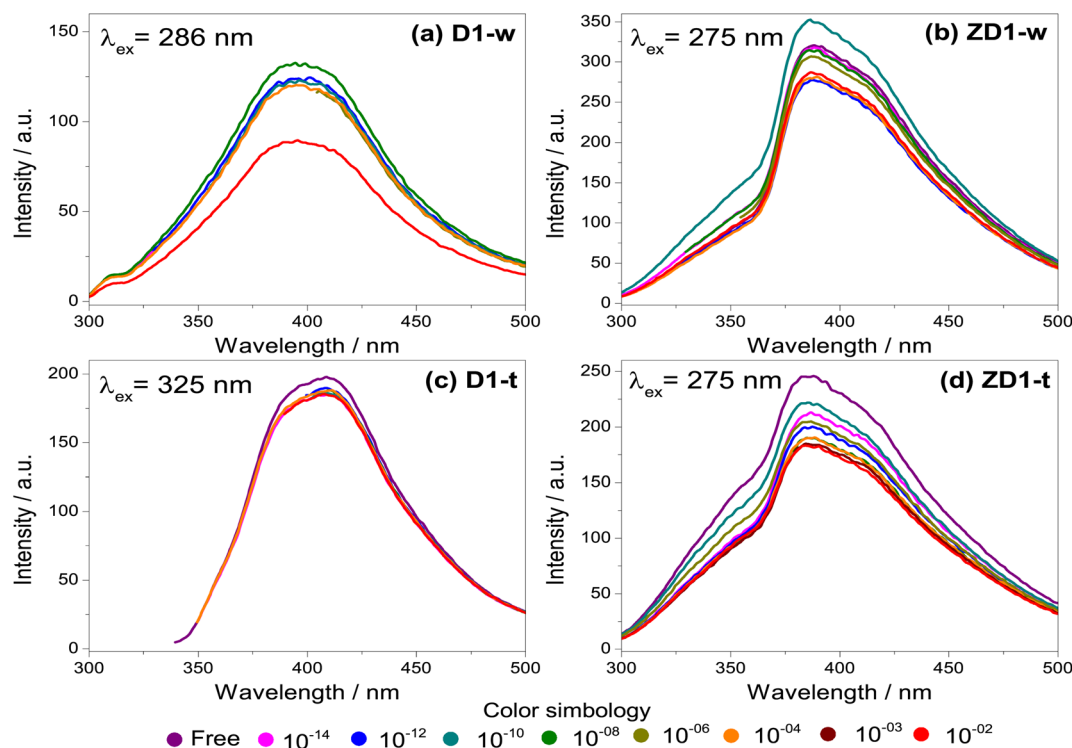


Fig. 5 Emission spectra for the H_2O_2 detection. The spectra were measured in aqueous phase for the samples without thermal treatment (a) D1-w and (b) ZD1-w and for the thermal treated (c) D1-t and (d) ZD1-t. For each sample the λ_{ex} which produced the maximum emission spectrum, was selected. The different concentrations of H_2O_2 are identified by the coloured dots marked at the bottom of the graphs and they are varied from free of H_2O_2 (fresh samples), up to a concentration of 10^{-2} M.

Table 2 Characteristic and intensities ratios for the Raman bands identified in Fig. 6

Sample	Band characteristics ^a												Ratio intensities				
	D1		D		D2		D3		G		D'		<i>I</i> _D / <i>I</i> _G	<i>I</i> _{D1} / <i>I</i> _G	<i>I</i> _{D2} / <i>I</i> _G	<i>I</i> _{D3} / <i>I</i> _G	<i>I</i> _{D'} / <i>I</i> _G
	p	w	p	w	p	w	p	w	p	w	p	w					
ZD1-t	1303	39	1345	59	1402	94	1520	139	1627	91	1677	64	0.92	0.37	1.21	1.50	0.41
D1-t	1312	49	1353	59	1407	89	1517	119	1610	86	1662	69	1.03	0.71	1.20	1.38	1.01
D2-t	1312	49	1355	63	1414	91	1516	120	1615	83	1664	66	1.17	0.64	1.17	1.52	0.74
D3-w	1310	48	1351	62	1404	83	1489	98	1565	100	1647	95	1.40	0.82	1.32	1.23	2.22

^a p = position of band peak in cm^{−1}; w = full width at half maximum (FWHM) in cm^{−1}.

^a p = position of band peak in cm^{-1} ; w = full width at half maximum (FWHM) in cm^{-1} .

the intensity, as specifically that shown in Fig. 5b, and (2) the presence of zinc contributes to improve the chemical stability of GQD.

3.4 Analysis of structure and inspection of the GQD morphology

3.4.1 Raman spectroscopy. With the aim to study the modifications resulted in the graphene oxide from the hydrothermal synthesis, Raman spectroscopy was achieved and results are shown in Fig. 6; also the bands position, full width at half maximum and intensities ratios (as I_D/I_G ratio, which is a parameter to measure the structure “disorder”⁴¹) are exhibited in the Table 2.

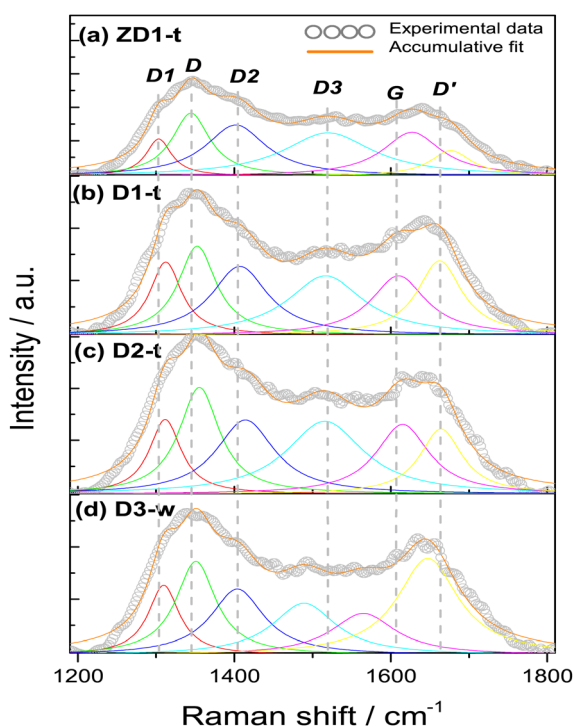


Fig. 6 Raman spectra from samples (a) ZD1-t, (b) D1-t, (c) D2-t and (d) D3-w. All spectra were taken on aqueous phase using a laser of $\lambda_{\text{ex}} = 785 \text{ nm}$. The deconvolution produced 6 bands modelled by the Lorentz equation and its accumulative fit the data with a coefficient $R^2 = 0.98$. The deconvolution represent the vibrations modes from the bands D1 = red, D = green, D2 = blue, D3 = cyan, G = pink, D' = yellow.

All the samples show the characteristics D, G and D' bands encountered in several research works.^{42–44} The D band around 1350 cm^{-1} , which is assigned to sp^3 hybridization from a disordered structure, it is found for simple graphene but most commonly in graphene oxide. Occasionally, the intensity of this band is employed to differentiate some carbon structures and it has been established that the intensity in the oxidizing samples is greater, due to the functionalization with oxygen groups where the amount of sp^3 hybridization increases;⁴⁵ this finding reinforces the slight increment in the FWHM in D2-t and D3-w, respective D1-t and ZD1-t. If this D band is associated to oxidation of graphene and ultimately to formation of GQD, it is worth to note that the I_D/I_G ratio for D3-w, is the highest with 1.40 (Table 2). This indicates that the particles conceived by the pretreatment of H-GOx, are more susceptible to be oxidized. Moreover, this might reflects that the epoxy groups, which are associated with defects onto the basal plane,⁴⁶ are responsible for the weakness and the split into smaller particles.^{41,47} In D1-t the value of 1.03 for I_D/I_G , is in accordance with the NH_4OH weakness to cause a substantial rupture in the GOx sheets. Notwithstanding in this process GQD were produced, but mainly, also conduced the amine groups to bond at the GQD structure. Owing to a similar value in the I_D/I_G ratio for ZD1-t, it is inferred that the ZnNP produced only a slight modification in the GQD structure, probably acquiring a more ordered character due to a chemical stability granted by the ZnNP. The D2-t is among D1-t and D3-w with $I_D/I_G = 1.17$ meaning that this structure was susceptible to be attacked by oxygen ions which increased its defects.

The G band which is related with a crystalline structure from graphene, is normally presented around 1580 cm^{-1} .⁴¹ Theoretically and experimentally it is known that shifts in the position and changes in the FWHM of the G band, are related with graphene layers. A shift to lower wavenumbers and an increment in FWHM reveals a trend to generate a graphene multilayer structure.^{48,49} Analysing the positions and the FWHM of the G bands for all the samples it is inferred that for D3-w, which shift to lower wavenumber and show an increment in FWHM, tend to form a multilayer structure contrary to D1-t, D2-t and ZD1-t (Table 2). The G band shift in D2-t is owing to it was synthesized with a lower NH_4OH concentration, which reduces the placement of amine groups onto the basal plane from the GQD; in the same way, the removal of NH_4OH leads to a lesser attraction among the GQD planes and this inhibits the



formation of layers. For ZD1-t there might be occurring a similar situation as in D2-t. As mentioned, the amine groups from GQD in the sample D1-t interact with ZnNP and in this way despite the FWHM grows the band shift to the right, suggesting that the attraction between GQD is null; this provoke the trend to produce thin layers (monolayers-like) mainly for those GQD in the aqueous solution (or not incorporated to ZnNP). Nevertheless, the close chemical interaction between the ZnNP and the GQD is indispensable, owing to neither NH_4OH nor ZnNP have strong photoluminescent properties (Fig. S1 and S2†).

The vibration modes from adjacent D1, D2 and D3 bands are related to formation of the edge states in GQD during its synthesis. For instance, the D1 at 1260 cm^{-1} is assigned to the functional group COOH , the D2 at 1435 cm^{-1} band is attributed to the presence of functional groups $\text{C}=\text{O}$,⁴⁷ whilst the D3 band at 1515 cm^{-1} , in this work, it was assigned to $\text{C}-\text{O}$ bond situated at the edges from GQD. When the samples are being synthesized the presence of water might induces the incorporation of COOH . This is the reason that the FWHM of D1 band is very similar in all samples and the position shifts to higher wave-numbers, highlighting the characteristic deformation of the graphitic structure and promotion of sp^3 hybridization. Notwithstanding, the slight decreasing in the FWHM of the D1 band for D3-w is in line with its FWHM values in the D2 and D3 bands; the FWHM of the D2 band for D3-w is lower than the rest of samples, however the rest have not excessively high values which indicates that all have $\text{C}=\text{O}$ groups at the edges. If it is compared the narrower FWHM and the shift to lower wave-numbers of the D3 band for D3-w, it can be inferred that it was more susceptible to be oxidized than D1-t and D2-t. Moreover, since $I_{\text{D1}}/I_{\text{G}}$ and $I_{\text{D2}}/I_{\text{G}}$ ratios for D3-w increased, these functional groups ($\text{C}=\text{O}$, COOH) are more abundant at the edges of this sample (Table 2). In fact, in the UV-Vis spectra (Fig. 3) the bands assigned to $\text{C}=\text{O}$ are higher for the non-treated samples than for those treated. Nevertheless, the amine groups exist in D3-w due to it exhibits photoluminescent properties (Fig. 4d). In D1-t and D2-t, the $I_{\text{D1}}/I_{\text{G}}$ and $I_{\text{D2}}/I_{\text{G}}$ ratios, are lower than D3-w and this indicates that the amine groups were bonded more easily at the edges which is also corroborated with the decreasing of the FWHM in D3 band for D3-w; the amines bonding was marked in D2-t as it shows lower values in this instance. The values of the $I_{\text{D1}}/I_{\text{G}}$ and $I_{\text{D2}}/I_{\text{G}}$ ratios for ZD1-t are similar to D1-t but particularly in the $I_{\text{D2}}/I_{\text{G}}$ ratio confirming the slight modifications promoted by the ZnNP. These modifications might be the chemical stability gained by GQD because of the $\text{C}=\text{O}$ and principally the COO functional groups losses. Analysing the $I_{\text{D3}}/I_{\text{G}}$ ratio it is observed that D1-t is amid D2-t and D3-w in this instance. While in D1-t the amine incorporation was straightforward in D3-w, with the lower $I_{\text{D3}}/I_{\text{G}}$ value, emphasizes the lack of amine groups at the edges (Table 2). As mentioned the trend from D3-w was to create multilayer of GQD; precisely the amine groups are responsible in the formation of this multilayer structure which for D3-w, act as ligand over the GQD planes, whilst there is not attraction at the edges between the same oxygen groups. Is reasonable the existence of epoxy and hydroxyl groups or ultimately (although less likely) π orbitals, all onto the GQD basal plane, in order to

attract the amine groups. Therefore it is likely that in D3-w and moderately in D1-t, the amine groups principally are placed at the basal plane and in D1-t also at the edges. It is noteworthy that although the GQD from D1-t and D3-w start to stack in multilayer, there is only a mildly growth as is discussed in the PL studies (section 3.3) and as is observed in the micrograph from Fig. 7a.

The D' band is related with the defects where the carbons tend to form pentagons or octagons instead of hexagons, besides to be associated with a graphitic structure.⁵⁰ The formation of this type of defects is directly related with the aggressive synthesis conditions during the synthesis of D1-t and D3-w which present the higher $I_{\text{D}'} / I_{\text{G}}$ ratios. Also the wide FWHM of D3-w and in less extent for D1-t might indicates higher predisposition to incorporate functional groups at the basal plane. On the contrary the rest of samples are prone to be attacked at the edges of the GQD. Moreover, D2-t and ZD1-t maintained its structure stable to chemical reaction induced

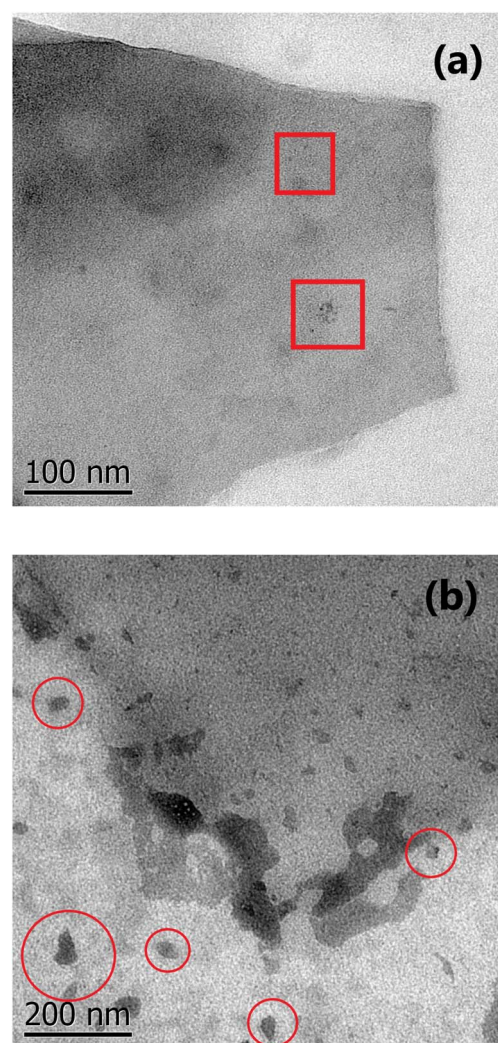


Fig. 7 TEM images from (a) D1-t: the red squares inside the micrography enclose the accumulated GQD very close between them and (b) ZD1-t: the red circles enclose the ZnNP which are mostly lesser than 100 nm.

by the ammonium hydroxide during the synthesis and due to the chemical stability by the ZnNP presence.

3.4.2 Samples morphology. All the spectroscopic techniques used in this work, evidenced a transformation once the samples were synthesized. These modifications can be traced from the graphite to the GOx synthesis and then to GQD. Those transformations were registered with SEM and TEM images, which are presented in Fig. S3† and 7, respectively. The SEM images from the Fig. S3a,† shows the graphite precursor which is composed with stacked particles with average size higher than 1 μm . The Fig. S3c,† shows the GOx particles and inside the red square can be observed particles which sizes are lesser than 1 μm . The magnification of Fig. S3c,† shows that the GOx particles are greater than 100 nm however as it was expected, they are smaller than graphite (Fig. S3d†). A further decrease was also awaited for the GQD synthesis and this can be observed in Fig. S3e.† These particles tend to accumulate in circles (with diameter around 1 μm) from particles with sizes lower than 100 nm (Fig. S3f†). With the heating treatment and the removal of the excess of NH_4OH the tend to form circles is not completely inhibited due to a small circle might be appreciated where the particles are around 5 nm (Fig. 7a). In the Fig. 7b can be observed that the ZnNP are around 50 nm and the GQD might be mostly incorporated onto the ZnNP structure, as in this micrography, there is not evidence of the GQD circles.

4 Discussion

4.1 GQD development and behaviour

According to the HT synthesis and to the characterization techniques used in this work, a scheme of GQD formation is presented in Fig. 8.

The principal course of action in D2-t was the defects formation owing to the low concentration of NH_4OH . These defects in D2-t were exploited by a slightly oxidant environment during the synthesis, maintaining epoxy groups at the basal plane and C–O at the edges (see the higher $I_{\text{D}3}/I_{\text{G}}$ ratio in Table 2 for D2-t). The lower NH_4OH concentration place some amines at the edges (substituting C=O and mainly COO^{-1} groups) which with the presence of C–O from adjacent GQD sheets, slightly and laterally increased the GQD plane. This indicates that the development of GQD in D2-t is given by the weakness from the carboxyl which are attacked by amine groups incorporating them at the edges (see $I_{\text{D}1}/I_{\text{G}}$ in Table 2 for D2-t and Fig. 8). On the other hand, the higher concentration of NH_4OH in D1-t assures that not only a structure rupture to occur, but also facilitates the placement of amine groups both, at the basal plane as at the edges. Similarly, the band related to C=O in Raman studies (see $I_{\text{D}2}/I_{\text{G}}$ in Table 2) in D1-t and D3-w has a high ratio; thus the amines are not bonded at this sites but they prefer the basal plane. Since in D1-t and D3-w amine groups are principally at the basal plane, the attraction is given in a vertical fashion and ultimately producing a multilayer structure specially in D3-w (Fig. 8). Such situations, lateral growth and multilayer structure are responsible of the slightly PL variations.

The excess of NH_4OH also varies the size of the GQD clusters, as it is shown in the SEM and TEM images from Fig. S3f† and 7a, and also influences the photoluminescent properties. This situation, is similar as that presented when higher concentrations of GOx are used to evaluate the PL spectra; a shift to lower energies is observed because the accumulation of some GOx particles (Fig. S4†). The influence of NH_4OH not only is limited to the structure rupture, availability of amine groups and the circle-shape of GQD, but also independently of the route, it can chemically reduce the GOx even though a low NH_4OH concentration was used (Fig. S5†).

4.2 GQD and ZnNP interaction

Photoluminescence studies (Fig. 5 and S4†) proved that slightly changes in the particles forming clusters might influence on the photoluminescent properties; changes in particle size can also impact on the interaction between GQD and ZnNP and finally with the H_2O_2 detection. Precisely, when NH_4OH is in excess stacking or lateral growth of the GQD might be occurring. For the detection of H_2O_2 with photoluminescence spectroscopy, initially is prepared a mixture of this oxidant and the samples containing GQD; just at the beginning, particles from GQD with size around 5 nm (fragments of GQD) with sp^2 domains, have a PL emission lower than the H_2O_2 free spectrum. After 5 minutes, these fragments creates clusters of sp^2 domains contacting sp^3 carbon structures starting an alteration in the intensity; this alteration is the growth of the spectrum, for instance for the 10^{-10} M H_2O_2 solution, higher than the H_2O_2 free PL spectrum (see Fig. 5b). Due to the amine groups there is an increase in the particles size, this is the contact of sp^2 – sp^3 which reduces the emission intensity with the course of time as recombination of small clusters.⁴⁰ If the particle size increases the band gap decreases and this might be also presenting in the GOx (Fig. S4†). As was proved, in spite that GQD poses photoluminescent properties, they do not detect H_2O_2 properly. When ZnNP is incorporated (GQD–ZnO nanocomposite) the fragments of GQD are attracted by the ZnO surface; principally where the oxygen from ZnO can trap the amine groups. The ZnO confer the GQD chemical stability avoiding the unnecessary reduction; in other words, the small clusters now are not in contact with this carbon-oxygen matrix with sp^3 hybridization. Moreover, the removal of NH_4OH enhances this interaction between ZnO and GQD, and now, only the spectra for the H_2O_2 concentrations of 10^{-14} and 10^{-12} (Fig. 5c and d) are not properly performed. However the for the rest of solutions concentrations, a gradual diminishing of the intensity is registered.

4.3 Steps on the H_2O_2 detection

In order to prove the involved steps for the H_2O_2 detection, a simple alternative test was carried out. For this test, the wasted solid from D1-w and H_2O_2 were lead to react and thus compare if it had a similar effect as in the H-GOx synthesis. A scheme concerning this experiment is shown in Fig. S6.† After HT reaction (between the wasted solid and H_2O_2), the liquid was filtered and named sD1-w. It was found in this test, that the



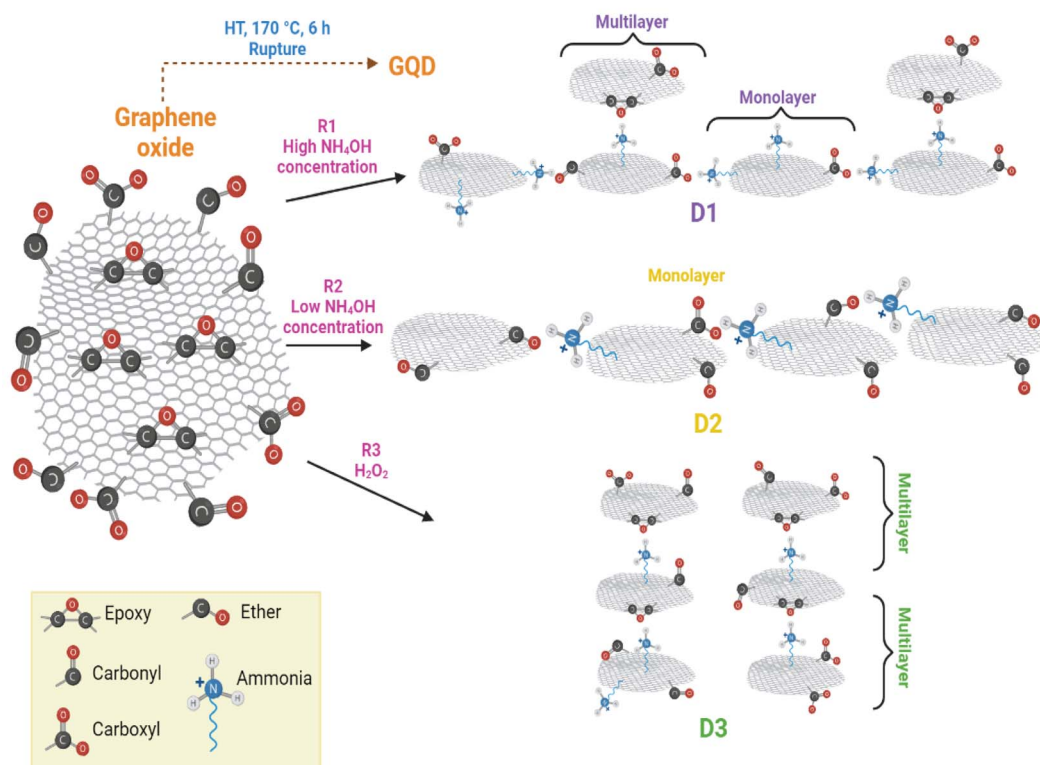


Fig. 8 Reaction scheme for the GQD formation by the three different routes described in section 2.3.

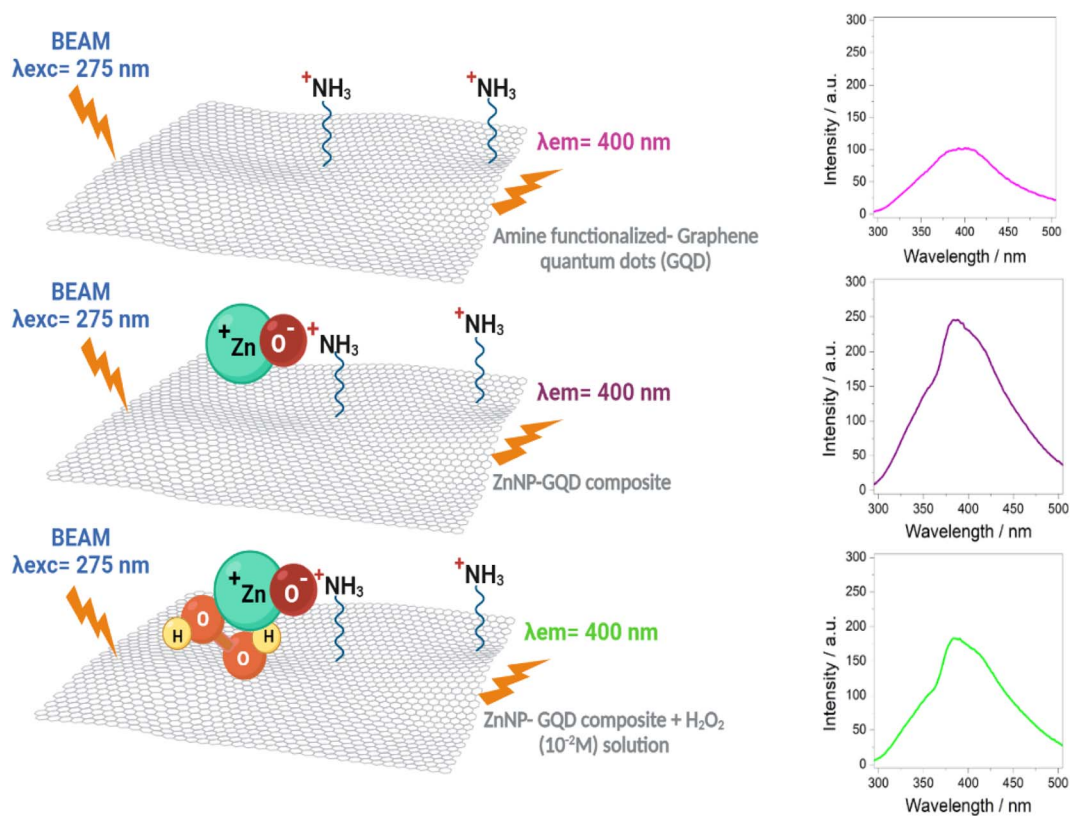


Fig. 9 Reaction scheme for the H_2O_2 detection in the ZD1-t sample shown in the Fig. 5d.

residual solid was slightly dissolved. The PL spectra (Fig. S7†) compares sD1-w with D1-w and H-GOx. The peaks intensities in sD1-w and D1-w are very similar and both are higher than H-GOx, although the position from sD1-w it is very similar to H-GOx. However if the PL spectrum from Fig. 4d, is compared with the sD1-w the λ_{ex} in both spectra is different, suggesting that H_2O_2 only induces the rupture from the solids but the PL emission is caused from the amine functional group. Therefore, due to the previous amines groups incorporation in D1-w and because the wasted solid from D1-w was not dissolved, it is inferred that once the amine groups are as ligands into the GQD structure it is very difficult to remove them. Even though a minor hydroxyls might be linked to GQD structure due to the peaks positions in sD1-w and H-GOx are similar. This proves that the followed steps for the detection are given *via* the ZnNP and H_2O_2 interaction. The presence of H_2O_2 might induces a slight polarization in the ZnNP structure perturbing the environment from the amine and the GQD. In the Fig. 9 a scheme from this reaction is given. In this way, the quenching in the H_2O_2 detection it is easier when its concentration is augmented, likely perturbing the polarity of the ZnO structure. In fact, it is feasible that the presence of H_2O_2 oxidize the ZnO owing to its high reduction standard potential. Finally, the quenching caused by the H_2O_2 over the ZD1-t particles can be described by fit the results with a quadratic equation; this equation represents the tendency of diminishing in the emission intensity as the oxidant concentration is increased and it is properly accomplished for the interval of H_2O_2 concentrations from 0.008 to 0.1 M (Fig. S8†).

5 Conclusions

The photoluminescence (PL) technique was sensible to GQD functionalized with amines. PL spectra suggested that GQD from sample D1-t (synthesized with high NH_4OH concentration) in general, resulted with lower particle size (emission peak centred at 406 nm) than D2-t synthesized with low NH_4OH concentration (emission peak centred at 418 nm). GQD produced with H_2O_2 in general were also smaller, however they presented weak PL emission intensity, until they were functionalized with amine groups (sample D3-w), indicating that emission was due to this ligand.

Because the $I_{\text{D3}}/I_{\text{G}}$ ratios concerning C–O groups at the edges of GQD in D1-t and D3-t are lower than in D2-t the amines groups were incorporated at the basal planes in smaller GQD particles while higher particle size tend to place amines at the edges.

The ZnO nanoparticles in the GQD (ZnO–GQD, nanocomposite) enhance the PL emission intensity. Since once the amine groups are bonded with GQD cannot be replaced by hydroxyl or oxygen ions, the H_2O_2 detection tested by photoluminescence studies, is achieved thanks to the interaction between ZnO nanocomposite and H_2O_2 , producing a quenching effect. This quenching is accentuated by the increasing of the H_2O_2 concentration and this behaviour can be described properly with a quadratic equation covering a concentration range from 0.008 M to 0.1 M.

Author contributions

Rolando Ramírez: conceptualization, methodology, investigation, writing original draft. Sara Rodríguez: writing review and editing. Genoveva Hernández: investigation and resources. Idalia Gómez: conceptualization, supervision, resources and project administration.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 K. S. Novoselov, V. I. Fal'ko, L. Colombo, P. R. Gellert, M. G. Schwab and K. Kim, *Nature*, 2012, **490**, 192–200.
- 2 R. Kaimal, V. Vinoth, A. S. Salunke, H. Valdés, R. V. Mangalaraja, B. Aljafari and S. Anandan, *Ultrason. Sonochem.*, 2022, **82**, 105868.
- 3 L. A. Ponomarenko, F. Schedin, M. I. Katsnelson, R. Yang, E. W. Hill, K. S. Novoselov and A. K. Geim, *Science*, 2008, **320**, 356–358.
- 4 H. Cao, W. Qi, X. Gao, Q. Wu, L. Tian and W. Wu, *Nanotheranostics*, 2022, **6**, 205–214.
- 5 A.-M. Croitoru, A. Moroşan, B. Tihăuan, O. Oprea, L. Moteliică, R. Truşcă, A. I. Nicoară, R.-C. Popescu, D. Savu, D. E. Mihăiescu and A. Ficai, *Nanomaterials*, 2022, **12**, 1943.
- 6 M. Facure, R. Schneider, L. Mercante and D. Correa, *Mater. Today Chem.*, 2022, **23**, 100755.
- 7 S. Akbari, *Int. J. Environ. Anal. Chem.*, 2020, **102**, 789–803.
- 8 R. Tian, S. Zhong, J. Wu, W. Jiang, Y. Shen, W. Jiang and T. Wang, *Opt. Mater.*, 2016, **60**, 204–208.
- 9 Y. R. Kumar, K. Deshmukh, M. M. N. Ali, G. Abhijay, W. A. Al-Onazi, A. M. Al-Mohaimed and S. K. Pasha, *Environ. Res.*, 2022, **203**, 111842.
- 10 S. M. Mousavi, S. A. Hashemi, A. Gholami, S. Mazraedost, W.-H. Chiang, O. Arjmand, N. Omidifar and A. Babapoor, *J. Sens.*, 2021, **2021**, 1–14.
- 11 P. Szustakiewicz, N. Kołsut, A. Leniart and W. Lewandowski, *Nanomaterials*, 2019, **9**, 602.
- 12 M. L. Yola and N. Atar, *J. Electrochem. Soc.*, 2016, **163**, B718–B725.



- 13 G. Chaloeipote, J. Samarnwong, P. Traiwatcharanon, T. Kerdcharoen and C. Wongchoosuk, *R. Soc. Open Sci.*, 2021, **8**, 210407.
- 14 T. T. B. Quyen, N. N. T. My, D. T. Pham and D. V. H. Thien, *Talanta Open*, 2022, **5**, 100103.
- 15 J. Wen, D. He, T. Zhuo, W. Li, J. Liu, J. Wu, Y. Zhao and Y. Yuan, *J. Environ. Chem. Eng.*, 2023, **11**, 109333.
- 16 Y. Zhang, Y. Liu, F. Sun and N. Yang, *Appl. Surf. Sci.*, 2023, **611**, 155655.
- 17 Z. Zhang, H. Liu, L. Zhai, J. Wu and L. Li, *Chem. Phys. Lett.*, 2023, **811**, 140177.
- 18 A. S. Siddiqui, M. A. Ahmad, M. H. Nawaz, A. Hayat and M. Nasir, *Catalysts*, 2022, **12**, 1105.
- 19 H. Shen, H. Liu and X. Wang, *Appl. Surf. Sci.*, 2023, **612**, 155816.
- 20 Z. Tan, C. Zhu, L. Han, X. Liao and C. Wang, *Sens. Actuators, B*, 2022, **373**, 132770.
- 21 Z. Xu, C. Zeng, Y. Zhao, M. Zhou, T. Lv, C. Song, T. Qin, L. Wang, B. Liu and X. Peng, *Food Chem.*, 2023, **410**, 135381.
- 22 M.-J. Lee, J.-A. Song, J.-H. Choi, J.-H. Shin, J.-W. Myeong, K.-P. Lee, T. Kim, K.-E. Park and B.-K. Oh, *Biosensors*, 2023, **13**, 289.
- 23 X. Deng, H. Tang and J. Jiang, *Anal. Bioanal. Chem.*, 2014, **406**, 6903–6916.
- 24 O. Adegoke, S. Khene and T. Nyokong, *J. Fluoresc.*, 2013, **23**, 963–974.
- 25 X. Liu, H. Tian, L. Yang, Y. Su, M. Guo and X. Song, *Sens. Actuators, B*, 2018, **255**, 1160–1165.
- 26 Y. He, X. Wang, J. Sun, S. Jiao, H. Chen, F. Gao and L. Wang, *Anal. Chim. Acta*, 2014, **810**, 71–78.
- 27 N. Yadav, V. Kallur, D. Chakraborty, P. Johari and B. Lochab, *ACS Omega*, 2019, **4**, 9407–9418.
- 28 S. Hembacher, F. J. Giessibl, J. Mannhart and C. F. Quate, *Proc. Natl. Acad. Sci.*, 2003, **100**, 12539–12542.
- 29 K. Xuan Sheng, Y. Xi Xu, C. Li and G. Quan Shi, *New Carbon Mater.*, 2011, **26**, 9–15.
- 30 W. Muhammad, N. Ullah, M. Haroon and B. H. Abbasi, *RSC Adv.*, 2019, **9**, 29541–29548.
- 31 Q. Lu, C. Wu, D. Liu, H. Wang, W. Su, H. Li, Y. Zhang and S. Yao, *Green Chem.*, 2017, **19**, 900–904.
- 32 X. Gao, Z. Ma, M. Sun, X. Liu, K. Zhong, L. Tang, X. Li and J. Li, *Food Chem.*, 2022, **369**, 130964.
- 33 D. L. Pavia, *Introduction to spectroscopy*, Harcourt College Publishers, 2001, p. 579.
- 34 Z. Benzait and L. Trabzon, *ACS Omega*, 2022, **7**, 37885–37895.
- 35 S. Kang, H. Han, K. Lee and K. M. Kim, *ACS Omega*, 2022, **7**, 2074–2081.
- 36 S. Talam, S. R. Karumuri and N. Gunnam, *ISRN Nanotechnol.*, 2012, **2012**, 1–6.
- 37 P. Zheng and N. Wu, *Chem.-Asian J.*, 2017, **12**, 2343–2353.
- 38 J. Robertson and E. P. O'Reilly, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1987, **35**, 2946–2957.
- 39 S. K. Cushing, M. Li, F. Huang and N. Wu, *ACS Nano*, 2013, **8**, 1002–1013.
- 40 G. Eda, Y.-Y. Lin, C. Mattevi, H. Yamaguchi, H.-A. Chen, I.-S. Chen, C.-W. Chen and M. Chhowalla, *Adv. Mater.*, 2010, **22**, 505–509.
- 41 L.-M. Peng, Z. Zhang and S. Wang, *Mater. Today*, 2014, **17**, 433–442.
- 42 S. H. Song, M.-H. Jang, J. Chung, S. H. Jin, B. H. Kim, S.-H. Hur, S. Yoo, Y.-H. Cho and S. Jeon, *Adv. Opt. Mater.*, 2014, **2**, 1016–1023.
- 43 E. Dervishi, Z. Ji, H. Htoon, M. Sykora and S. K. Doorn, *Nanoscale*, 2019, **11**, 16571–16581.
- 44 D. Liu, X. Chen, Y. Hu, T. Sun, Z. Song, Y. Zheng, Y. Cao, Z. Cai, M. Cao, L. Peng, Y. Huang, L. Du, W. Yang, G. Chen, D. Wei, A. T. S. Wee and D. Wei, *Nat. Commun.*, 2018, **9**, 1–10.
- 45 M. Pumera, *Nanoscale Res. Lett.*, 2007, **2**, 87–93.
- 46 X. Gao, J. Jang and S. Nagase, *J. Phys. Chem. C*, 2009, **114**, 832–842.
- 47 G. Rajender and P. K. Giri, *J. Mater. Chem. C*, 2016, **4**, 10852–10865.
- 48 R. Vuppuladhadiam, H. E. Jackson and R. L. C. Wu, *J. Appl. Phys.*, 1995, **77**, 2714–2718.
- 49 H. Wang, Y. Wang, X. Cao, M. Feng and G. Lan, *J. Raman Spectrosc.*, 2009, **40**, 1791–1796.
- 50 O. Vryonis, T. Andritsch, A. S. Vaughan and P. L. Lewin, *J. Mater. Sci.*, 2019, **54**, 8302–8318.

