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Site-selective functionalization of plasmonic nanopores for enhanced fluorescence emission rate and Förster resonance energy transfer†

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In this work, we use a site-selective functionalization strategy to decorate plasmonic nanopores with fluorescent dyes. Using an easy and robust fabrication method, we manage to build plasmonic rings on top of dielectric nanotubes with different inner diameters. The modulation of the dimension of the nanopores allows us to tailor their field confinement and their Purcell factor in the visible spectral range. In order to investigate how the changes in geometry influence the fluorescence emission rate efficiency, thiol-conjugated dyes are anchored on the plasmonic ring, thus forming a functional nanopore. We study the lifetime of ATTO 520 and ATTO 590 attached in two different configurations: single dye and FRET pair. For the single dye configuration, we observe that the lifetime of both single dyes decreases as the size of the nanopore is reduced. The smallest nanopores yield an experimental Purcell factor of 6. For the FRET pair configuration, we measure two regimes. For large nanopore sizes, the FRET efficiency remains constant. Whereas for smaller sizes, the FRET efficiency increases from 30 up to 45% with a decrease of the nanopore size. These findings, which have been supported by numerical simulations, may open new perspectives towards energy transfer engineering in plasmonic nanopores with potential applications in photonics and biosensing, in particular in single-molecule detection and sequencing.

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Introduction

In recent years, plasmonic nanopores have been proposed for applications in single molecule detection towards sequencing.¹ Compared to the more common solid-state nanopores which are typically used for single molecule experiments based on ionic current measurements,² plasmonic nanopores offer interesting advantages for optical spectroscopic approaches.^{3–5} Some of the key features of plasmonic nanopores are reduction of detection volume, localization of the electromagnetic field and increase of the signal-to-noise ratio. As a result, plasmonic nanopores find application in single molecule detection and sequencing based on Surface Enhanced Raman Scattering (SERS),⁴ as well as in fluorescence spectroscopy experiments.^{6,7} In particular, enhanced fluorescence in plasmonic nanopores has been verified in single molecule DNA detection.^{3,8} In these studies, one or more nucleotides are tagged with a suitable dye, and their enhanced emission during the nanopore

translocation process enables the fingerprint recording of the molecule. It is worth mentioning that fingerprinting of a single molecule has been recently proposed for another important application, *i.e.* protein sequencing.⁹ In a recent article, van Ginkel and co-workers sequentially read out the signals from a single protein by means of the Förster Resonant Energy Transfer (FRET) between a dye fixed on a nanopore and the dyes attached to the protein.¹⁰ Despite not using a plasmonic nanopore, this proof-of-concept experiment shows the potentiality of using FRET to carry out single molecule sequencing using a nanopore. As far as FRET experiments using plasmonic nanostructures are concerned, in 2014 Ghenuche *et al.* demonstrated that a plasmonic nanohole can produce a six-fold increase in the single molecule FRET rate.¹¹ Furthermore, in 2016 de Torres *et al.* showed that plasmonic antennas can enable forbidden dipole–dipole Förster energy transfer exchanges.¹² Despite the significant growing interest in this topic, there have not been many studies on FRET experiments using plasmonic nanopores. In addition, no flow-through experiments making use of FRET effects have been reported so far. One of the key aspects to perform FRET-based sequencing experiments is the immobilization of one of the FRET pair dyes close to a plasmonic surface. In this respect, surface functionalization protocols for dye grafting have been extensively investigated,¹³ but the selective functionalization at the specific positions where the plasmonic nanostructure is present is still

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challenging and just a few studies have been reported on that.^{14–16} Note that proper site-selective functionalization significantly reduces the amount of the background signal coming from the substrate as well as facilitating the investigation of fluorescence emission from specific isolated points. Clearly, fundamental studies to optimize the use of plasmonic nanopores are still missing.

In this work, we show that it is possible to achieve an easy site-selective decoration of metallic nanopores in large arrays with thiol-modified dyes. Although we are not able to control the number of molecules, our functionalization method has great control over the nanoplasmonic sites to be functionalized, and it is done in one single step. We use this functionalization method to experimentally characterize the emission rate of plasmonic nanopores of four different sizes. The characterization is carried out for single fluorophore and donor/acceptor (D/A) FRET configurations. The experimental results are supported by numerical simulations. In particular, we show that smaller nanopores produce higher field enhancement and higher fluorescence emission rates. We observe that the D/A pairs have a FRET efficiency of 30% for large nanopores, and we find that this efficiency increases when the size of the nanopore shrinks. We believe that these configurations can find important applications in single molecule flow-through experiments for single molecule detection towards sequencing.

Results and discussion

We have fabricated plasmonic nanopores by building a gold ring on top of a dielectric hollow pillar. The fabrication procedure, described in the Methods section, allows us to obtain different diameters of the hole, spanning from a few tens to hundreds of nm. Representative SEM images of illustrative samples are displayed in Fig. 1. In our experiments, four different geometries have been investigated, each of them corresponding to a different ionic current used for the fabrication (see Methods). The pillars have been prepared in 30×30 micron arrays, with a distance of 5 micron between adjacent pillars. The four representative samples, defined by the dimensions of their inner (D_{in}) and outer (D_{out}) diameters of the plasmonic holes are given in Table 1.

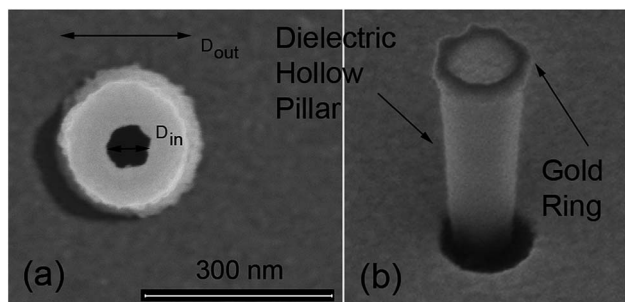


Fig. 1 SEM images of the fabricated nanopores. (a) Nanoring prepared with 80 pA ion current. (b) Tilted view of the dielectric pillar and the metallic ring at the top. The thickness of the gold nanorings is 30 nm (see Methods).

Table 1 Notation and average dimensions of the four different kinds of fabricated nanopores

Notation	D_{in} (nm)	D_{out} (nm)
A	70 ± 30	230 ± 30
B	170 ± 30	310 ± 40
C	250 ± 60	370 ± 10
D	330 ± 40	500 ± 10

We expect that different diameters correspond to different configurations of field confinement and enhancement inside the nanoholes. Finite Element Methods (FEM) simulations implemented in Comsol Multiphysics with the Radio Frequency Module have been used in order to elucidate the behaviour of the realized nanopores. In Fig. 2(a), we show the spectra of the near-field enhancement for the four structures, calculated under a linearly polarized and tightly focused Gaussian beam. It is observed that the spectrum of the nanopores redshifts as the inner diameter is increased. This fact will be specially relevant for the FRET experiments shown at the end of the manuscript, as some studies have shown that the spectral position of the resonance of the structure affects the FRET efficiency of the dyes in the vicinity of the structure.¹⁷ It is also seen that the average field enhancement in the ring volume is enhanced as the size is decreased. Then, in Fig. 2(b), we show the average of the field intensity enhancement $\langle |E/E_0|^2 \rangle$ calculated in the whole volume of the nanopore (see Methods).

The illumination in Fig. 2(b) is chosen to be the same as in the experiments, *i.e.* a tightly focused Gaussian beam at 532 nm (see Methods for more details). We observe that the field enhancement that the dyes feel inside the pore decreases when the size of the nanopore is increased. In addition, looking at the near-field profiles in the insets of Fig. 2(b), it is observed that the smallest nanopore ($D_{\text{in}} = 70$ nm) is the only one where the field is confined in the whole volume of the pore. In the other cases, the field is especially localized around the walls of the nanopore. This fact is in agreement with some previous results on similar structures.¹⁸ For the sake of completeness, we have also computed the scattering cross section of the four kinds of nanopores. The results are given in the ESI.†

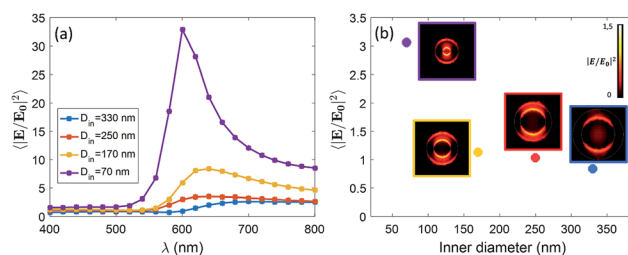


Fig. 2 (a) Spectra of the intensity enhancement for the four nanopores under the illumination of a tightly focused Gaussian beam. (b) Average field intensity enhancement for the four different fabricated nanopores calculated in the volume of the plasmonic ring. The simulations are done for a linearly polarized Gaussian beam that is tightly focused ($w_0 = 350$ nm) at 532 nm. Insets: near-field profiles induced by the tightly focused Gaussian beam at 532 nm.



The fabricated nanostructures comprise an isolated gold ring on top of a dielectric tube, as can be seen in Fig. 1 and 3(a). The choice of this configuration/geometry is indeed the key aspect for the site-selective functionalization method used here. The nanotube is used as a nanochannel through which molecules can translocate/diffuse. The proposed method, inspired by a functionalization strategy recently reported for 2D materials,¹⁹ is based on the conjugation between a gold (or another noble metal) surface and a thiolate-dye (Fig. 3). In particular, to perform a site selective decoration of metallic holes, we used a HS-PEG2000-NH₂ molecule as a linker between the gold surface and NHS-activated ATTO dyes. Note that the PEG2000 molecule has a size of approximately 6 nm and lower linker lengths can be achieved using shorter PEG molecules.²⁰ While the protocol of functionalization of the dyes is reported in Methods, the procedure of nanopore decoration is the following (see Fig. 3(a)):

(1) A 50 μ M solution of HS-PEG-NH-OC-ATTO dye is prepared in EtOH. (2) The plasmonic nanopores are prepared on a Si₃N₄ membrane. The metal is only deposited on one side of the substrate (see Methods). (3) The sample is cleaned in oxygen plasma for 180 seconds to facilitate the diffusion of the dyes through the nanochannels. The surface that is cleaned is the one where there is no metal. (4) We suspend the sample on top of an EtOH bath. The metallic face of the sample is in contact

with the EtOH bath, whereas the Si₃N₄ face is in contact with air. (5) We wet the dry side with a 3 μ L droplet of HS-PEG-NH-OC-ATTO diluted in EtOH. Due to the different concentrations, the dyes move towards the EtOH bath and they get attached to the metallic part when they try to reach it. (6) Before the droplet containing the solution with dyes starts to dry off, we add another droplet of 3 μ L of the same solution. We repeat this process three times. (7) The sample is rinsed off in a second pure EtOH bath, and the site-selective functionalization is achieved. It is worth mentioning that we chose EtOH in order to increase the wettability of the nanopores.

In order to avoid background contributions, we are interested in functionalizing plasmonic nanopores isolated from the substrate. However, this method of site-selective functionalization can be used on any nanostructure that contains nanochannels. The thiol-terminated dye strategy is suitable for noble metals, while other functional groups can be defined in order to apply this method also to other materials.¹³ In Fig. 3(b) and (c), we depict fluorescence confocal images from two different samples functionalized with ATTO 520 and ATTO 590 respectively. Both images have been taken with a Nikon A1 confocal microscope. In Fig. 3(b) and (c) we show confocal fluorescence images obtained in the 500–550 nm (570–620 nm) range using a laser at 488 nm (561 nm). It can be observed that the fluorescence is localized over the array of nanopores, hence demonstrating correct site-selective functionalization. Moreover, an additional experiment was performed to show that the signal stemming from non-specific adsorption of the dyes without the –SH was negligible. We also see that the fluorescence intensity is not the same for all the nanopores. This is because our protocol does not allow us to control the number of molecules that get attached on the metallic surface. This is not an issue for this study, as the lifetime measurements that we have performed are independent of the number of molecules. That is, different concentrations of dyes yield the same lifetimes. Thus, we leave the control of the number of molecules attached during the functionalization process as future work.

After verifying that our site-selective functionalization method works properly, we set out to study the emission properties of ATTO 520 and ATTO 590 dyes. As mentioned earlier, we have measured the emission properties of the dyes in two different functionalization configurations: single dye and FRET (D/A) pair. For each of the two functionalization configurations, we have measured four different arrays of antennas. Each array is composed of 49 nanopores, each of them having a different size. The mean internal/external diameter (D_{in}/D_{out}) has been given in Table 1.

The lifetime measurements have been carried out using a pulsed laser at 532 nm. We have used two channels of detection. One channel is set up to detect the fluorescence from the ATTO 520 dye in the 553–577 nm spectral band. The other channel detects the fluorescence emitted in between 593 and 643 nm, which mainly corresponds to the emission of ATTO 590. The protocol that we have followed to measure the lifetime of the ATTO dyes as well as the details about the optical set-up are given in Methods. In Fig. 4(c), we have displayed representative measurements of the lifetime traces that we have obtained for the

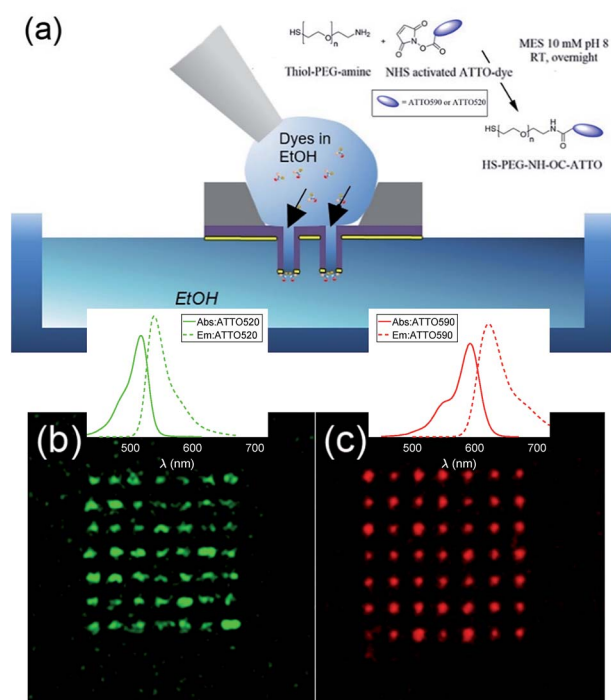


Fig. 3 (a) Sketch of the site-selective functionalization process. The nanopores are in contact with an EtOH bath. The other face of the sample is in contact with a solution of dyes in EtOH. The dyes diffuse through the channels and get attached to the nanopore thanks to the thiol group. Fluorescence images of plasmonic nanopore sites selectively decorated with ATTO dyes. (b) The nanopores are functionalized with ATTO 520. The laser excitation is at 488 nm and the fluorescence detection is at 500–550 nm. (c) The nanopores are functionalized with ATTO 590. The laser excitation is at 561 nm and the fluorescence detection is at 570–620 nm.



four structures functionalized with ATTO 520 dye. Moreover, a reference lifetime measurement of ATTO 520 dye thiol-functionalized on top of the gold flat surface of the membrane has also been included. In Fig. 4(a), we display the lifetime measurements that have been obtained after functionalizing the four different fabricated arrays of nanopores with single ATTO 520 and 590 dyes. Each data point in Fig. 4(a) is the result of averaging out approximately 100 lifetime measurements (see Methods). It is observed that both dyes follow a similar lifetime *vs.* nanopore size trend. That is, the lifetime of both dyes decreases as the inner diameter of the nanopores shrinks.

This phenomenon is quite intuitive: when the size of the nanopores is smaller, it is more likely that the emission of the dyes gets reflected off the inner walls of the nanopore and absorbed back by the emitter, thus enhancing the emission rate. Note that this intuitive picture has to do with the semi-classical interpretation of the enhancement of decay rates²¹ and it is not linked to the field computations in Fig. 2. In contrast, understanding why the lifetime reduction is more drastic for ATTO 590 than for ATTO 520 is less intuitive. We have carried out some simulations to see how the emission of an electric dipole emitting at 565 nm (ATTO 520) and at 618 nm (ATTO 590) is like. We have placed the dipole at 6 nm of the inner wall of the nanopore (to account for the PEG molecule), and computed the intensity inside the nanopore (see Methods). Given a unitary dipolar emission, we expect the emission to increase as the size of the pore is decreased, since the emission goes as the inverse of the lifetime.²¹ This behaviour is

indeed well captured by the simulations in Fig. 4(b). Moreover, the near-field simulations also capture the big change in the slope of the trend line that the ATTO 590 dye follows in between 170 and 250 nm. The insets in the Fig. 4(b) panel are the near-field distributions induced by the dipole excitation for each of the four structures. For the sake of completeness, in the Fig. 4(d) panel, we have displayed the far-field intensity distributions as insets, and their corresponding integrals on a plane. We see that the same general behaviour (emission decreases as size increases) mostly holds. Note that Fig. 4(b) is quantitatively different from Fig. 2(b). In Fig. 2(b), the near-field is obtained under the illumination of a tightly focused Gaussian beam. The Gaussian beam preserves the symmetry of the system and it only has propagating components. In contrast, as mentioned above, the near-field plotted in Fig. 4(b) is obtained with an emitting dipole. The emitting dipole attached to the inner surface of the nanopore breaks the symmetry of the structure. In addition, the emitting dipole has both evanescent and propagating components. Consequently, the nanopore creates a different near-field distribution for the two different excitations. In order to get an idea of the mean Purcell factor (or enhancement of the spontaneous decay rate Γ/Γ_0) produced by the functionalized nanopore, we have measured the lifetime of the HS-PEG-NH-OC-ATTO dyes deposited on top of the gold-coated membrane. That is, the dyes are spotted and then anchored in the same membrane as the plasmonic nanopores, but out of their influence. In Table 2, we present the results of these measurements, as well as the corresponding Purcell factors for all the nanopores and the two dyes. It is observed that the lifetime of both dyes on top of a golden membrane is of the order of 3.5 ns. Given these lifetimes, the nanopores yield Purcell factors in between 1 and 6, which are in accordance with those measured for similar structures.²² Here, it is important to stress a couple of facts. First, fluorescence is the convolution of absorption and emission, and therefore an increased emission rate (or Purcell factor greater than 1) does not imply that the fluorescence signal will be increased. Second, the spectral shape of the Purcell factor²³ may be completely different from the spectral shape of the intensity enhancement. Thus, choosing an excitation wavelength according to the intensity maxima shown in Fig. 2 would not necessarily yield a greater lifetime reduction. Finally, and for the sake of completeness, the lifetimes of the ATTO dyes have also been measured on top a glass coverslip, yielding (3.9 ± 0.2) ns and (4.3 ± 0.3) ns for ATTO 520 and ATTO 590 respectively.

We have also characterized the nanopores in the FRET configuration. That is, we have attached both ATTO dyes at the inner wall of the nanopores using our site-selective functionalization. The concentration used for the functionalization has been halved with respect to the concentration used for the single dyes. Note that the FRET effects that are measured here are average effects, as the distance between the D/A pairs cannot be controlled in an efficient manner. In Fig. 5(a), we depict the lifetime of the ATTO 520 dye both in single and in FRET configuration. In Fig. 5(b), we plot the FRET efficiency obtained using the data in Fig. 5(a). The efficiency is obtained as $\eta_{\text{FRET}} = 1 - \tau_{\text{DA}}/\tau_{\text{D}}$ (see Methods). These two plots give us different information. Firstly, in Fig. 5(a) we observed that the presence

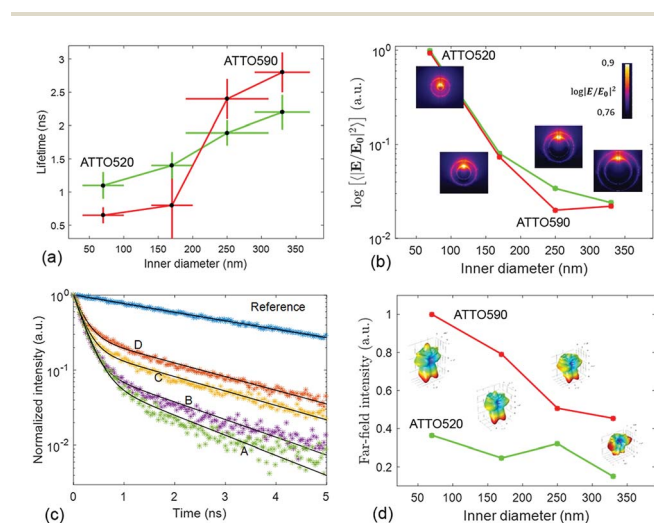


Fig. 4 (a) Lifetime measurements of site-selective functionalized plasmonic nanopores with single ATTO 520 and ATTO 590 as a function of the inner diameter of the nanopore. Each data point is the result of the measurement of two arrays of 49 nanopores (see Methods). The fluorescence from the ATTO 520 (590) dye has been measured in the 553–577 (593–643) nm spectral band. (b) Average intensity emitted by an electric dipole in the nanopore volume. The dipoles emit at 565 nm and 618 nm respectively. The insets are near-field distributions of the field induced by the dipole emitting at 565 nm. (c) Representative lifetime measurements of the four nanopores, as well as the reference case. (d) Far-field intensity emitted by dipoles emitting at 565 nm and 618 nm. The far-field is calculated on a horizontal plane on top of the nanopore. The insets are the far-field distributions induced by the emitting dipole at 618 nm.



Table 2 Lifetime measurements of ATTO dyes dried off on a membrane

Dye	Lifetime (ns)	I/I_0 for A	I/I_0 for B	I/I_0 for C	I/I_0 for D
ATTO 520	3.5 ± 0.2	3.2 ± 0.6	2.5 ± 0.4	1.9 ± 0.2	1.6 ± 0.2
ATTO 590	3.6 ± 0.1	6 ± 1	5 ± 3	1.5 ± 0.2	1.3 ± 0.1

of the acceptor shortens the lifetime of the donor, which is one of the trademark effects of FRET interaction.

Then, in Fig. 5(b) we see that the FRET efficiency is almost constant for large nanopore sizes (30% and 31%), and then it increases as the size of the nanopore shrinks. Note that despite the large error bars, this trend is followed by the mean value of η_{FRET} as well as by $(\eta_{\text{FRET}} + \delta\eta_{\text{FRET}})$ and $(\eta_{\text{FRET}} - \delta\eta_{\text{FRET}})$. Also, note that even if the distance between the dyes is not constant, the efficiency values obtained are comparable to those obtained by some other groups using similar structures.²⁴ The efficiency *vs.* size behaviour can also be intuitively explained: we assume that the dyes are attached onto the wall of the nanopore at a constant rate. That is, we assume that each infinitesimal area of the nanopore has the same density of dyes. When the nanopore is large, we assume that the FRET efficiency stems from the interaction between the dyes in this infinitesimal area. That is, the dyes interact with their first neighbours, but they do not interact with the dyes in the other infinitesimal areas, as the distance between them is longer than the typical 1–10 nm FRET range. This situation changes as the curvature of the inner wall increases. The dyes from each infinitesimal area effectively

approached the rest of the neighbouring areas. As a result, the interaction between D/A pairs is no longer restricted to first neighbours, thus increasing the efficiency of the FRET energy exchange. If we consider that each infinitesimal area of the inner wall can be modelled by an effective mean D/A pair, the increase of the curvature can be seen as a reduction of the distances between neighbouring pairs. A sketch of this explanation is depicted in Fig. 6, where an orthogonal line to the surface of the inner wall determines the position of the dyes on the PEG molecule. In this toy model, it is observed that the average distance between neighbouring FRET pairs is reduced from 7.4 to 6.7 nm due to the increase of the curvature of the inner wall. This change in average distance is quite consistent with the FRET radius that we have experimentally obtained, which is 7.5 ± 0.9 and 6.8 ± 0.8 nm for the largest and smallest nanopore respectively (see Methods). That is, our functionalization method allows us to attach two dyes in the inner walls of the nanopores, and then approximately retrieve the curvature of the nanopore *via* lifetime measurements. Our results are in accordance with some other studies where D/A pairs exchanged energy under the influence of a resonant structure.¹⁷ Zhao *et al.* showed that the energy exchange increased its efficiency when the resonance of the structure was spectrally close to the absorption of the acceptor. As shown in Fig. 2(a), the resonance of the smallest nanopores is at 600 nm, which is spectrally close to the absorption maximum for the ATTO 590 dye, which is at 590 nm. Last but not least, note that an alternative way of measuring the FRET efficiency is *via* acceptor sensitization.²⁵ However, we have not been able to do it as the fluorescence signal decreased due to photobleaching (see Methods).

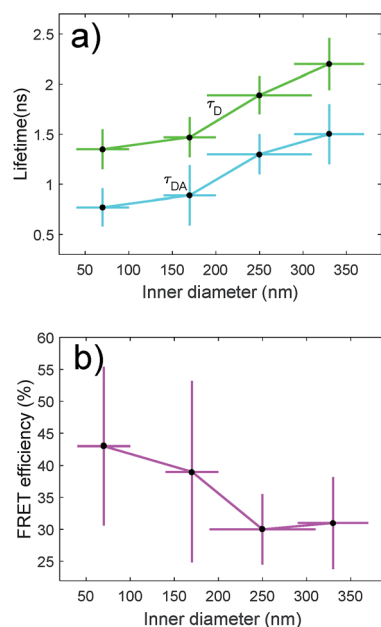


Fig. 5 (a) Lifetime measurements of site-selective functionalized plasmonic nanopores with single ATTO 520 and with ATTO 520 and ATTO 590 altogether. Each data point is the result of the measurement of two arrays of 49 nanopores (see Methods). The fluorescence from the ATTO 520 dye has been measured in the 553–577 nm spectral band. (b) Average FRET efficiency for the ATTO 520–ATTO 590 FRET pair. The FRET efficiency is computed with the relation $\eta_{\text{FRET}} = 1 - \tau_{\text{DA}}/\tau_{\text{D}}$ with the data given in (a).

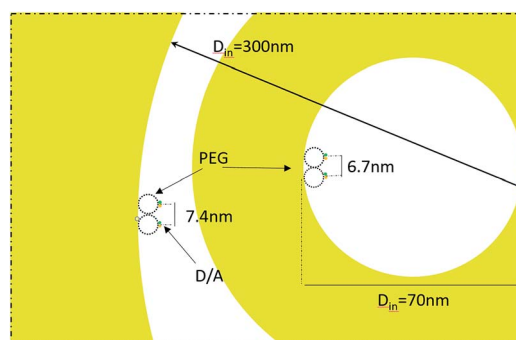


Fig. 6 Sketch of the curvature influence on the FRET efficiency. The structure in the centre of the picture represents a nanopore with an inner diameter of $D_{\text{in}} = 70$ nm. The outer yellow represents a nanopore with an inner diameter of $D_{\text{in}} = 300$ nm. The PEG molecules are displayed as dashed circles of 6 nm. The ATTO dyes are drawn as coloured dots at the end of the PEG molecules. An orthogonal line to the surface of the inner wall determines the position of the dyes on the PEG molecule.



Conclusions

In summary, we have investigated a site-selective functionalization technique to decorate metallic nanopores. The nanopores have been fabricated using a robust technique that allows us to control the dimension of the inner channel of the nanopore. In order to functionalize the metallic nanopores, we have modified NHS-activated dyes with a thiolate PEG. We have attached the HS-PEG-NH-OC-ATTO dyes to the plasmonic ring using a concentration gradient in between the two sides of the sample. We have functionalized the structures for four different nanopores sizes. We have observed that the emission rate of the plasmonic nanostructures increases as the inner diameter of the nanopore shrinks. This trend has been tested for single dyes as well as for FRET pairs. In the case of FRET pairs, it has been observed that the efficiency of the energy exchange increases for small sizes, whereas it stays constant for large sizes. The novel approach herein presented, in which plasmonic nanopores are used for fluorescence energy transfer, may lead to robust detection schemes for single molecule detection in flow-through configuration, which is one of the best options for carrying out sequencing experiments.²⁶

Methods

Nanopore array fabrication

The fabrication of the metallic nanoholes follows simple and robust procedures. The substrate is a Si_3N_4 membrane (100 nm thick) prepared on a silicon chip. The fabrication of the plasmonic ring nanopore array follows the procedure illustrated in ref. 27 and used by our group in several recent papers. A thin layer of S1813 optical resist is spun on top of the membrane with a final thickness equal to the height of the structure we want to obtain. This layer is exposed by secondary electrons during the FIB milling of the nanoholes from the bottom of the membrane. For this experiment, four ionic currents have been used: 80, 230, 430 and 790 pA. Smaller radii can be achieved by using a lower ionic current still preserving the robustness of the procedure. After the exposure, the samples have been developed in acetone and rinsed in EtOH. Finally, e-beam depositions (*ca.* 3 nm Ti and 30 nm Au) have been performed in order to deposit the ring on top of the obtained dielectric pillar. The directionality of the physical evaporation process is fundamental to avoid the deposition along the pillar wall. A deposition rate below $0.3 \text{ \AA s}^{-1}$ ensures low roughness. All the samples have been treated in O_2 plasma before the dye functionalization.

Optical set-up

The experiments are performed on an inverted microscope with a long distance water-immersion microscope objective with $\text{NA} = 1.1$. The light beam, which enters the microscope through its rear port, is obtained with a fibred supercontinuum laser with a central wavelength of $\lambda = 532 \text{ nm}$. The spectral bandwidth of the laser is $\pm 5 \text{ nm}$, the repetition rate is 78 MHz and the pulse width is approximately 65 ps. The power of the laser before entering the rear port of the microscope is $0.35 \text{ \mu W}$. The samples that we use for the experiment are held on a sample

holder, which is attached to a micro and a nanopositioner. The nanopositioner is used to center the nanopore with respect to the incident beam. We consider that the nanopore is centered with respect to the beam when the fluorescence intensity is maximum. The nanopositioner is also used to place the sample at the z plane where the fluorescence counts are maximized. The detection of the fluorescence is done in reflection. The fluorescence is separated from the backscattering of the sample by means of three filters. Firstly, a notch dichroic filter at 532 nm and a longpass filter at 550 nm remove most of the backscattering. Then, a dichroic filter at 594 nm splits the fluorescence in two paths. The fluorescence on the path for wavelengths below 594 nm mostly belongs to the ATTO 520, while the fluorescence for wavelengths longer than 594 nm is mostly due to the emission of ATTO 590. A different third filter is added on each path. A bandpass filter at 565/24 nm (618/50 nm) is added on the path of detection of ATTO 520 (ATTO 590). An avalanche photodiode (APDs) is set up on each path to detect the fluorescence. The signal of the APDs is recorded with a time-correlated single photon counting module.

Lifetime measurements

The APDs as well as the laser are connected to a time-correlated single-photon counting module (Swabian Instruments) in time-resolved mode. Making use of a home-built code, we build a histogram of 300 bins, each of them having a temporal width of 30 ps. A time delay of 6.56 ns is applied to the laser channel, so that the histogram monotonously decreases. The histogram measurement is carried out for 25 s, and it is repeated four times, yielding four histograms. A biexponential function of the kind $Ae^{-t/\tau_A} + Be^{-t/\tau_B}$ is used to fit the decay curve for each histogram. Typically, the Ae^{-t/τ_A} exponential contains information about the backscattering (or impulse response function), whereas Be^{-t/τ_B} contains information about the fluorescence. This is easily observed, as the fluorescence signal tends to decrease over time due to the photobleaching of dyes which are illuminated for long periods close to a metallic surface. This photobleaching effect is usually captured in the fit by a decrease of the coefficient B . Then, measuring four histograms allows us to see whether the lifetime measurement is stable over time or not. We obtain the lifetime of a particular nanopore by doing a weighted arithmetic mean of the four lifetimes, the weights being B/A . The same measurement is carried out for each of the nanopores of the array (49 in total), and it is repeated for a second array. Then, the lifetime of the array is given as the average of all the significant lifetimes of the two arrays. The lifetime measurements that are not considered significant are those whose values are more than 2σ away from the average. The error of the measurement is obtained as the standard deviation associated with the average of the significant points.

FRET measurements

The FRET efficiency of the antennas is computed as $\eta_{\text{FRET}} = 1 - \tau_{\text{DA}}/\tau_{\text{D}}$, with τ_{D} being the lifetime of the donor by itself and τ_{DA} the lifetime of the donor when it is in the neighborhood of an acceptor. Due to the fact that we cannot control the distances in



between the donor and the acceptor, our experiments yield mean measurements of the FRET efficiency. The concentration of the donor has been maintained for both experiments, *i.e.* single donor and FRET pair. A control check with acceptor sensitization has not been performed because of our lack of control of the number of molecules and the decreasing fluorescence signal due to photobleaching. The error of η_{FRET} is computed by performing a standard analysis of the statistical

propagation of the uncertainty: $s_f = \sqrt{\sum_i \left(\frac{\partial f}{\partial x_i} \right)^2 s_{x_i}^2}$.

The D/A mean distance is obtained using the expression

$R = R_0 \left(\frac{\tau_{\text{DA}}}{\tau_{\text{D}} - \tau_{\text{DA}}} \right)^{1/6}$, where $R_0 = 6.5$ nm is the Förster distance of the two ATTO dyes used in the experiment.²⁵

Chemical modification of ATTO dyes

HS-PEG2000-NH₂ [mercapto poly(ethylene glycol) amine] (2.27 mg, 1135 μmol) was solubilized in 1 mL of MES 10 mM pH 8 (1.14 mM). From this stock solution, aliquots of 100 μM of HS-PEG2000-NH₂ were prepared using the same buffer, MES 10 mM pH 8.

100 μL of the HS-PEG2000-NH₂ aliquot was added to a solution of ATTO 590-NHS ester (100 μM) in MES 10 mM pH 8. The mixture was stirred at room temperature, in the dark, overnight. The same procedure was followed to label HS-PEG2000-NH₂ with ATTO 520-NHS ester. Even in such a case, the two reagents were used in the stoichiometric molar ratio of 10 : 1, respectively for the dye and PEG. The excess of dye was then removed by 24 hours of dialysis (cut-off membrane: 1 kDa). Once purified, the labelled PEG was lyophilised overnight (Lio5P, Kam-bic). The amount of dye effectively linked to PEG was quantified by measuring the absorbance in water of the labelled PEG and by means of the specific molar attenuation coefficient and Beer-Lambert law. The as-prepared HS-PEG-NH-OC-ATTO 590 and HS-PEG-NH-OC-ATTO 520 were stored at -20 °C in aqueous solution and diluted at the desired concentration in EtOH prior to surface functionalization experiments.

Materials. ATTO 520 NHS ester, BioReagent, suitable for fluorescence, $\geq 80.0\%$ (coupling to amines), ATTO 590 NHS ester, BioReagent, suitable for fluorescence, $\geq 60\%$ (coupling to amines), HS-PEG2000-NH₂ HCl salt (average Mn 2000) and 4-morpholineethanesulfonic acid (MES) hydrate, $\geq 99.5\%$, were purchased from Sigma Aldrich. MES 10 mM pH 8 was prepared as follows: a stock solution of 100 mM MES was prepared by solubilizing MES hydrate (9.76 g, 50 mmol) in DI water (500 mL); 10 mL of this solution was diluted with 90 mL of DI water reaching 10 mM as the final concentration; the pH was adjusted to pH 8 by adding 0.1 M NaOH dropwise. Spectrum™ Spectra/ Por® 6 Standard RC Pre-wetted Dialysis Tubing MWCP 1 kD was purchased from Fisher.

Electromagnetic simulations

Numerical simulations based on the Finite-Element Method implemented in the RF Module of Comsol Multiphysics® were carried out to investigate the electromagnetic response of an

isolated nanopore. The dimensions of the simulated structures were set according to the average sizes obtained from SEM measurements. A dielectric constant $n = 1.33$ was used for water, and $n = 1.5$ was set for the glass substrate. The refractive index of Au was taken from Rakic.²⁸ The model computes the electromagnetic field in each point of the simulation region, enabling the extraction of the quantities plotted along the manuscript. The unit cell was set to be 1500 nm wide in both x- and y-directions and 2000 nm along the z-direction, with perfect matching layers (200 nm thick) at the borders. A linearly polarized Gaussian beam is tightly focused down to a spot size of $w_0 = 350$ nm. The beam impinges on the top side of the structure, namely where the nanopore is placed, from the water side. The field intensity enhancement is computed as the average of field intensity in the volume of the nanopore. We define the volume as $\pi D_{\text{in}} t/4$, with $t = 30$ nm being the width of the deposited gold ring.

For the simulations with the dipoles, a dipole oscillating along the in-plane direction of the nanopore at either 565 nm or 618 nm was placed at 6 nm from the inner wall of the nanopore for all four cases treated in the manuscript. The orientation of the dipole is arbitrary, as we checked that its orientation did not modify the trend shown in Fig. 4(b). In this case the dipole is the only field source within the simulation region. The field intensity enhancement is also computed across the whole volume of the pore.

The far-field, *i.e.* the scattering amplitude of the structure in the momentum space, is calculated from the near field using the Stratton-Chu formula,²⁹ namely $E_{\text{far}} = \frac{ik}{4\pi} r_0 \times \int [n \times E - \eta r_0 \times (n \times H)] e^{ik \cdot r_0} dS$, where E and H are the fields on the surface S enclosing the structure, r is the radius vector (not a unit vector) of the surface S , E_{far} is the field calculated at the point r_0 , n is the unit normal to the surface S , k is the wave number and $\eta = \sqrt{\mu/\epsilon}$ is the impedance. Then, the far-field calculations carried out in Fig. 4(d) are computed as an average of the far-field intensity projected on a plane at infinity. The plane is located in the semi-space where the water-immersion objective is located.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- 1 A. B. Dahlin, *Analyst*, 2015, **140**, 4748–4759.
- 2 X. Shi, D. Verschuere, S. Pud and C. Dekker, *Small*, 2018, **14**, 1703307.
- 3 O. N. Assad, T. Gilboa, J. Spitzberg, M. Juhasz, E. Weinhold and A. Meller, *Adv. Mater.*, 2017, **29**, 1605442.



- 4 C. Chen, Y. Li, S. Kerman, P. Neutens, K. Willems, S. Cornelissen, L. Lagae, T. Stakenborg and P. Van Dorpe, *Nat. Commun.*, 2018, **9**, 1733.
- 5 G. A. Chansin, R. Mulero, J. Hong, M. J. Kim, A. J. DeMello and J. B. Edel, *Nano Lett.*, 2007, **7**, 2901–2906.
- 6 D. Punj, P. Ghenuche, S. B. Moparthy, J. de Torres, V. Grigoriev, H. Rigneault and J. Wenger, *Wiley Interdiscip. Rev.: Nanomed. Nanobiotechnol.*, 2014, **6**, 268–282.
- 7 P. Ponzellini, X. Zambrana-Puyalto, N. Maccaferri, L. Lanzanò, F. De Angelis and D. Garoli, *Nanoscale*, 2018, **10**, 17362–17369.
- 8 J. Larkin, R. Y. Henley, V. Jadhav, J. Korlach and M. Wanunu, *Nat. Nanotechnol.*, 2017, **1**–8.
- 9 L. Restrepo-Pérez, C. Joo and C. Dekker, *Nat. Nanotechnol.*, 2018, **13**, 786–796.
- 10 J. van Ginkel, M. Filius, M. Szczepaniak, P. Tulinski, A. S. Meyer and C. Joo, *Proc. Natl. Acad. Sci. U. S. A.*, 2018, 201707207.
- 11 P. Ghenuche, J. De Torres, S. B. Moparthy and V. Grigoriev, *Nano Lett.*, 2014, **14**, 4707–4714.
- 12 J. de Torres, M. Mivelle, S. B. Moparthy, H. Rigneault, N. F. Van Hulst, M. F. García-Parajó, E. Margeat and J. Wenger, *Nano Lett.*, 2016, **16**(10), 6222–6230.
- 13 M. Oliverio, S. Perotto, G. C. Messina, L. Lovato and F. De Angelis, *ACS Appl. Mater. Interfaces*, 2017, **9**, 29394–29411.
- 14 A. Ananth, M. Genua, N. Aissaoui, L. Díaz, N. B. Eisele, S. Frey, C. Dekker, R. P. Richter and D. Görlich, *Small*, 2018, **14**, 1703357.
- 15 M. Pla-Roca, L. Isa, K. Kumar and E. Reimhult, *ACS Appl. Mater. Interfaces*, 2015, **7**, 6030–6035.
- 16 M. P. Jonsson, A. B. Dahlin, L. Feuz, S. Petronis and F. Höök, *Anal. Chem.*, 2010, **82**, 2087–2094.
- 17 L. Zhao, T. Ming, L. Shao, H. Chen and J. Wang, *J. Phys. Chem. C*, 2012, **116**, 8287–8296.
- 18 J. Aizpurua, P. Hanarp, D. S. Sutherland, M. Käll, G. W. Bryant and F. J. García de Abajo, *Phys. Rev. Lett.*, 2003, **90**, 4.
- 19 D. Garoli, D. Mosconi, E. Miele, N. Maccaferri, M. Ardini, G. Giovannini, M. Dipalo and F. De Angelis, *Nanoscale*, 2018, **10**, 17105–17111.
- 20 M. Zhengyu, N. L. B. David, M. L. Sharon, A. S. Kim, L. K. Michael, E. D. Dennis and H. F. Terri, *PLoS One*, 2014, **9**, 1–10.
- 21 L. Novotny and B. Hecht, *Principles of Nano-Optics*, 2006.
- 22 J. De Torres, P. Ghenuche, S. B. Moparthy, V. Grigoriev and J. Wenger, *ChemPhysChem*, 2015, **16**, 782–788.
- 23 X. Zambrana-Puyalto and N. Bonod, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **91**, 195422.
- 24 D. J. Roth, M. Nasir, P. Ginzburg, P. Wang, A. Le Marois, K. Suhling, D. Richards and A. V. Zayats, *ACS Photonics*, 2018, **5**(11), 4594–4603.
- 25 I. Medintz and N. Hildebrandt, *FRET – Förster Resonance Energy Transfer*, 2013.
- 26 K. Lee, K. B. Park, H. J. Kim, J. S. Yu, H. Chae, H. M. Kim and K. B. Kim, *Adv. Mater.*, 2018, 1704680.
- 27 D. Garoli, E. Calandrini, A. Bozzola, M. Ortolani, S. Cattarin, S. Barison, A. Toma and F. De Angelis, *Nanoscale*, 2017, **9**, 915–922.
- 28 A. D. Rakic, A. B. Djuricic, J. M. Elazar and M. L. Majewski, *Appl. Opt.*, 1998, **37**, 5271–5283.
- 29 D. K. Cheng, *Field and Wave Electromagnetics*, 1989.

