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Reversing Mg suppression effect on Co-site water oxidation of MgCo_2O_4 based on vanadium-atom electronic affinity synergy with Mg sites toward electronic redistribution[†]

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Active-site regeneration from d-band-center control engineering and electronic redistribution based on target-atom implantation is of enormous importance for efficient water oxidation. Here, low-activity Co catalytic sites of MgCo_2O_4 are reactivated for large-current (500 mA cm^{-2}) and high-efficiency water oxidation through vanadium-atom electronic affinity synergy with Mg sites toward charge rearrangement around Co sites. Interestingly, very little vanadium (1.3 wt%) was implanted into MgCo_2O_4 to achieve atom-scale structure tailoring and electronic redistribution, reversing the Mg suppression effect on Co-site water oxidation of MgCo_2O_4 . Vanadium-implantation-tailored MgCo_2O_4 coupled with MgO ($\text{V}-\text{MgCo}_2\text{O}_4@\text{MgO}$) exhibits significantly enhanced oxygen evolution reaction performance, with low overpotentials of 240 and 290 mV for 100 and 500 mA cm^{-2} , respectively, in 1 M KOH. Operating at 500 mA cm^{-2} , $\text{V}-\text{MgCo}_2\text{O}_4@\text{MgO}$ has good catalytic stability for at least 20 hours. This work constructs excellent catalysts through atomic-level structural regulation and provides a new perspective for the principle of electronic-affinity synergy between V and Mg species toward Co-O bond optimization.

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1. Introduction

The increasing global energy crisis and serious environmental pollution have forced researchers to develop sustainable clean energy without secondary pollution.^{1–3} Hydrogen (H_2) as an ideal substitute for traditional fossil fuels with high energy density can be produced *via* electrochemical water splitting driven by electricity from wind or solar energy.^{4–6} The process involves the cathodic two-electron hydrogen evolution reaction (HER) and the anodic four-electron oxygen evolution reaction (OER), and serves as the most promising route among various H_2 -production strategies.^{7–9} The OER counterpart suffers from slow kinetics and high overpotentials, enormously limiting the efficiency of water splitting ($2\text{H}_2\text{O} \rightarrow 2\text{H}_2 + \text{O}_2$). Tremendous effort has been devoted to exploring stable and efficient electrocatalysts¹⁰ to facilitate water splitting. Noble-metal materials such as Pt, Ir

and Ru are still considered as the most advanced catalytic electrodes,^{11,12} but high cost and scarcity severely restrict their scaled-up utilization.

Non-precious-metal-based catalysts with advantages of large abundance, easy synthesis and low price were explored to catalyze water decomposition including oxides,¹³ phosphides,¹⁴ nitrides,¹⁵ hydroxides,¹⁶ etc. Among them, the AB_2O_4 -type spinel structure has attracted more attention whose tetrahedral and octahedral geometry can better accommodate various transition-metal cations^{17–20} for excellent structural flexibility. Cobalt-relevant spinel materials, *e.g.* NiCo_2O_4 ,²¹ CuCo_2O_4 ,²² MnCo_2O_4 ,¹⁹ FeCo_2O_4 ,²³ ZnCo_2O_4 ²² and LiCo_2O_4 ,²⁴ are widely utilized as water-splitting catalysts.

Spinel-type oxides formed by Mg and transition-metal elements often demonstrate poor electrocatalytic performance.^{25,26} For example, Kumar *et al.* reported that MgFe_2O_4 is a promising material for semiconductor photocatalytic water splitting, but its low electro-conductivity hinders the electrocatalytic application.²⁷ Even if changing Fe sites into the Co element toward MgCo_2O_4 , the as-synthesized urea- and (nitrilotriacetic acid)-stabilized MgCo_2O_4 catalysts for electrochemical water oxidation require overpotentials of 463 and 573 mV, respectively, for 10 mA cm^{-2} catalytic current density in 1 M KOH.²⁸ Indeed, associated electrochemical

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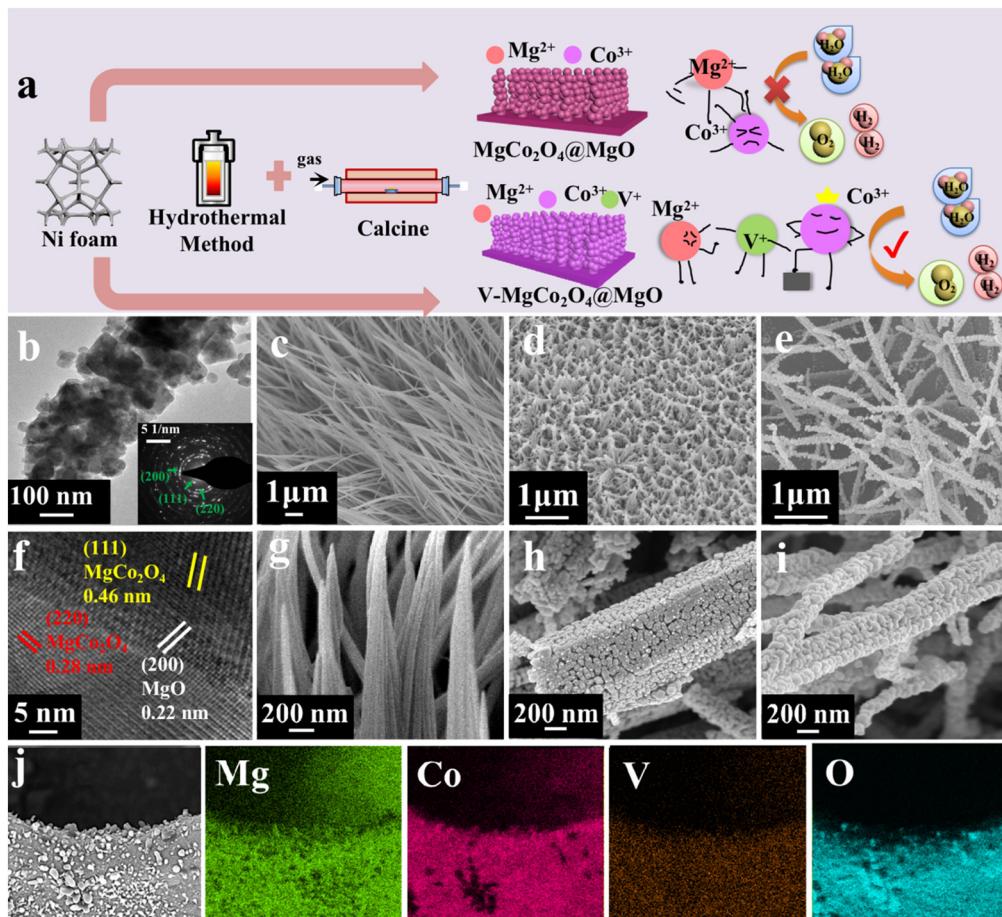


Fig. 1 Material preparation and characterization. (a) Schematic diagram for V-MgCo₂O₄@MgO and MgCo₂O₄@MgO preparation. (b) TEM image and SAED pattern of V-MgCo₂O₄@MgO. (f) HRTEM image of V-MgCo₂O₄@MgO. (c and g) V-CoOOH@MgCO₃, (d and h) V-MgCo₂O₄@MgO, (e and i) MgCo₂O₄@MgO SEM images. (j) EDX elemental mapping of Mg, Co, V and O for V-MgCo₂O₄@MgO.

performances are extremely unsatisfactory, necessitating abundant scientific input to promote fundamental understanding of catalytic activity enhancement.^{29,30} Qiu *et al.* constructed a MgCo₂O₄@WO₃ core–shell heterostructure by a simple one-step hydrothermal coordination for rapid electron transfer of MgCo₂O₄.³¹

The lower electronegativity of Mg in the Mg-relevant spinel structure may cause large electron density of adjacent transition-metal sites toward undesirable intermediate binding.³² Compared with Mg, however, the smaller atomic size and higher electron-affinity ability of V make it an intriguing atom which can be implanted into a spinel platform, destroying the lattice periodicity of the catalysts^{33,34} and balancing the electronic distribution. Notably, V-atom-utilization was also proven as a pivotal way to regulate the d-band center of active sites toward appropriate d-band intensity and bandwidth.^{35,36} Moreover, the synergistic protocol of V and Mg for electronic structure regulation of adjacent metal sites is also ambiguous and must be interesting.^{37,38}

Based on the above consideration, we achieved the co-occurrence of the vanadium and Mg atoms in the MgCo₂O₄

platform as V-MgCo₂O₄@MgO (vanadium-doping-tailored MgCo₂O₄ coupled with MgO) to influence the adjacent Co active sites for optimizing electronic structure. Unlike the low catalytic activity of the MgCo₂O₄ sample, the appearance of few V atoms (1.3 wt%) endows MgCo₂O₄ with superhigh water-oxidation activity and long-term large-current catalytic stability. V-MgCo₂O₄@MgO exhibits significantly reinforced OER performance, with low overpotentials of 240 and 290 mV for 100 and 500 mA cm⁻², respectively, in 1 M KOH. Operating at 500 mA cm⁻², V-MgCo₂O₄@MgO presents good catalytic stability for at least 20 hours.

2. Experimental

2.1 Materials

CoCl₂·6H₂O, MgCl₂·6H₂O, VCl₃ and urea were purchased from Aladdin Co Ltd. (China). All chemical reagents are used without further purification. Deionized water was manufactured with a Millipore system and used in all experiments.

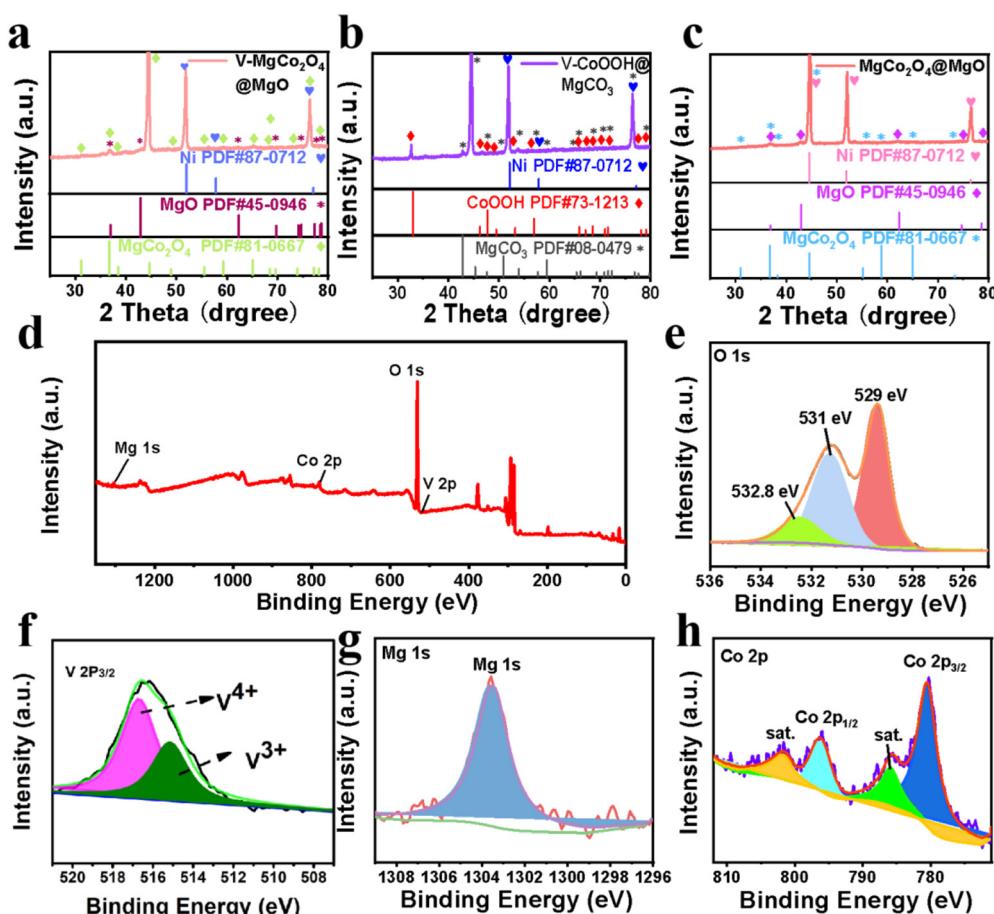


Fig. 2 XRD and XPS analyses. XRD spectra of (a) V-MgCo₂O₄@MgO, (b) V-CoOOH@MgCO₃ and (c) MgCo₂O₄@MgO. (d) XPS survey spectrum of V-MgCo₂O₄@MgO. XPS spectra of (e) O 1s, (f) V 2p, (g) Mg 1s, and (h) Co 2p for V-MgCo₂O₄@MgO.

2.2 Synthesis of V-MgCo₂O₄@MgO and MgCo₂O₄ on Ni foam

Ni foam (2 cm × 4 cm) was ultrasound-treated in 5 wt% hydrochloric acid for 10 minutes, and subsequently washed with distilled water and alcohol. CoCl₂·6H₂O (8 mmol), MgCl₂·6H₂O (4 mmol), VCl₃ (1 mmol) and urea (16 mmol) were dissolved in 30 mL water to form a light red solution. Then, the mixed solution was transferred to a Teflon-lined autoclave with NF immersed in the solution. The autoclave was sealed and kept at 120 °C for 10 h. The as-obtained V-CoOOH@MgCO₃ was pyrolyzed at 400 °C in an Ar atmosphere for 4 h to obtain V-MgCo₂O₄@MgO.

The synthesis process of CoOOH@MgCO₃ was similar to that of V-CoOOH@MgCO₃ without V species. CoOOH@MgCO₃ was annealed at 400 °C to obtain MgCo₂O₄@MgO.

2.3 Characterization

X-ray diffraction (XRD) data were collected on a LabX XRD-6100 X-ray diffractometer equipped with a Cu K α radiation source. X-ray photoelectron spectroscopy (XPS) results are recorded on an ESCALABMK II ray photoelectron spectrometer (Mg is the excitation source). Scanning electron microscopy (SEM) measurements were performed on a

Hitachi S-4800 field emission scanning electron microscope with an accelerating voltage of 20 kV. Transmission electron microscopy (TEM) measurements were carried out with a Zeiss Libra 200FE transmission electron microscope operated at 200 kV. Inductively coupled plasma optical emission spectrometry (ICP-OES) was performed on a PlasmaPro100 Cobra300, Oxford.

Electrochemical data were collected on a CHI 760e electrochemical analyzer with a scan rate of 5 mV s⁻¹ in 1 M KOH, utilizing the as-prepared sample (0.5 × 0.5 cm²) as the working electrode, platinum as the counter electrode, and Hg/HgO as the reference electrode. The equation of “ E vs. RHE = E vs. Hg/HgO + (0.098 + 0.0591 pH) V” was used to convert IR corrected LSV curves. The overpotential for the OER was obtained via $\eta = E$ vs. RHE - 1.23 V.

3. Results and discussion

Fig. 1a shows multi-step preparation processes of V-MgCo₂O₄@MgO and MgCo₂O₄@MgO, which are described in detail in the Experimental section. Combining the hydrothermal coordination strategy and calcination treatment, very few vanadium atoms entered the MgCo₂O₄@MgO system as V-MgCo₂O₄@MgO, alleviating the Mg suppression for Co-site

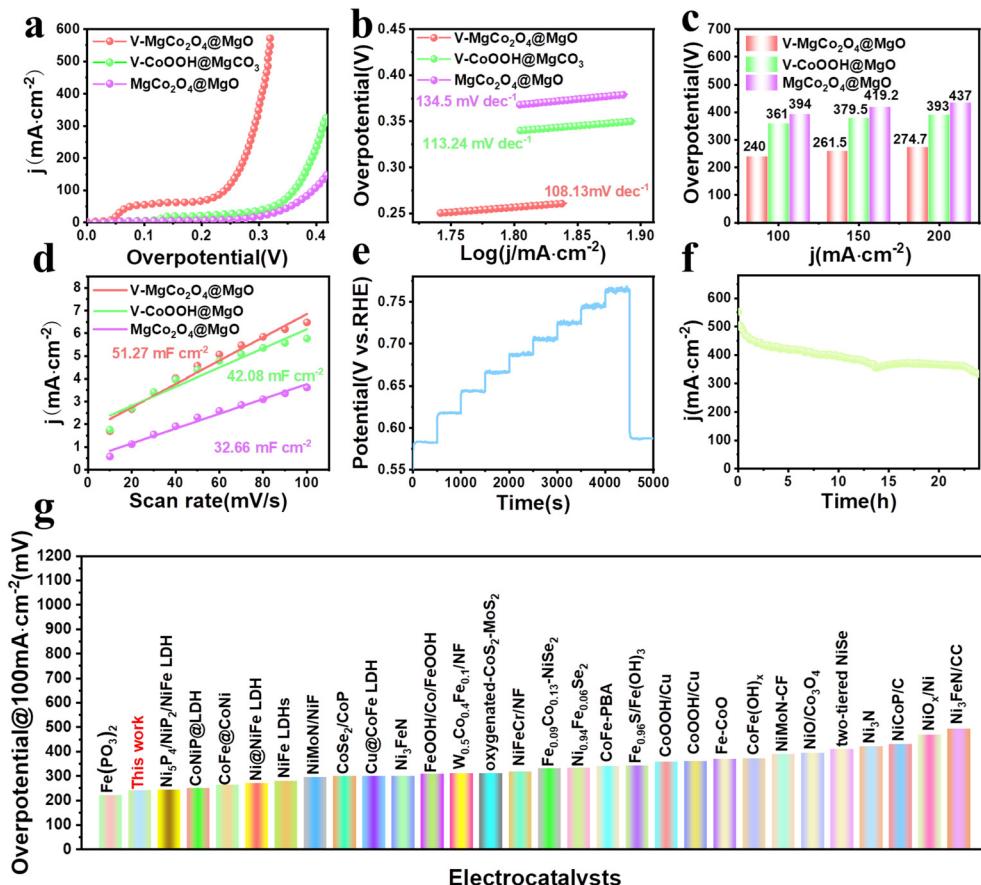


Fig. 3 Evaluation of OER performance. (a) LSV curves and (b) Tafel plots of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$, $\text{MgCo}_2\text{O}_4@\text{MgO}$ and $\text{V-CoOOH}@{\text{MgCO}_3}$ at a scan rate of 5 mV s^{-1} . (c) The required overpotentials for $\text{V-MgCo}_2\text{O}_4@\text{MgO}$, $\text{V-CoOOH}@{\text{MgCO}_3}$ and $\text{MgCo}_2\text{O}_4@\text{MgO}$ to drive 100 , 150 , and 200 mA cm^{-2} . (d) Linear plot of capacitive current dependent on scan rate for $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ ($R^2 = 0.9887$), $\text{V-CoOOH}@{\text{MgCO}_3}$ ($R^2 = 0.9856$) and $\text{MgCo}_2\text{O}_4@\text{MgO}$ ($R^2 = 0.9912$). (e) Multistep chronopotentiometry test for the $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ electrode in 1.0 M KOH without IR correction. The current density ranges from 10 mA cm^{-2} to 100 mA cm^{-2} , with an increment of 10 mA cm^{-2} per 500 s . (f) Time-dependent current density curve for $\text{V-MgCo}_2\text{O}_4@\text{MgO}$. (g) Comparison of overpotential for $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ at 100 mA cm^{-2} with other electrocatalysts recently reported in 1 M KOH .

water oxidation. Unlike the low water-oxidation activity of $\text{MgCo}_2\text{O}_4@\text{MgO}$, Co sites in $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ are activated with the optimized electronic structure and Co-O bond. $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ was characterized by transmission electron microscopy (TEM) and scanning electron microscopy (SEM). The TEM image in Fig. 1b reveals the rod-like morphology of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ constituted by nanoparticles (with a diameter *ca.* 60 nm), which is beneficial for active-site exposure and electrolyte permeation. The selected area electron diffraction (SAED) pattern of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ (inset in Fig. 1b) well displays typical diffraction rings corresponding to the (220) and (111) planes of spinel MgCo_2O_4 and the (200) plane of MgO , consistent with X-ray powder diffraction (XRD) results (Fig. 2a). The related high-resolution TEM (HRTEM) image of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ also reveals the existence of the (220) and (111) planes of spinel MgCo_2O_4 and the (200) plane of MgO (Fig. 1f) with plane spacings of 0.28 , 0.46 and 0.22 nm , respectively. For SEM characterization, $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ (Fig. 1d and h) demonstrates a rod-like feature with a rough surface. In

addition, the energy dispersive X-ray (EDX) element mapping is shown in Fig. 1j, confirming the elemental composition of Mg, Co, O and V. Abundant $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ was deposited on Ni foam (Fig. S1†). For comparison, the SEM and EDX element mapping images of $\text{MgCo}_2\text{O}_4@\text{MgO}$ and $\text{V-CoOOH}@{\text{MgCO}_3}$ are shown in Fig. S2.† As the precursor to prepare $\text{V-MgCo}_2\text{O}_4@\text{MgO}$, vanadium-containing $\text{CoOOH}@{\text{MgCO}_3}$ ($\text{V-CoOOH}@{\text{MgCO}_3}$) presents a smooth nanowire morphology (Fig. 1c and g), uniformly grown on nickel foam. $\text{MgCo}_2\text{O}_4@\text{MgO}$ (Fig. 1e and i) presents a curved nanowire morphology with a zigzag surface.

Fig. 2a and c show the XRD patterns of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ and $\text{MgCo}_2\text{O}_4@\text{MgO}$, respectively. For $\text{V-MgCo}_2\text{O}_4@\text{MgO}$, XRD characteristic peaks of MgCo_2O_4 (JCPDS NO. 81-0667) and MgO (JCPDS No. 45-0946) are detected, (Fig. 2a) which are similar to that of $\text{MgCo}_2\text{O}_4@\text{MgO}$ (Fig. 2c). In the $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ XRD curve, the characteristic signal of vanadium species was not found, indicating vanadium-atom doping into $\text{MgCo}_2\text{O}_4@\text{MgO}$ as $\text{V-MgCo}_2\text{O}_4@\text{MgO}$. The mass content of V in $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ is measured as 1.3 wt\%

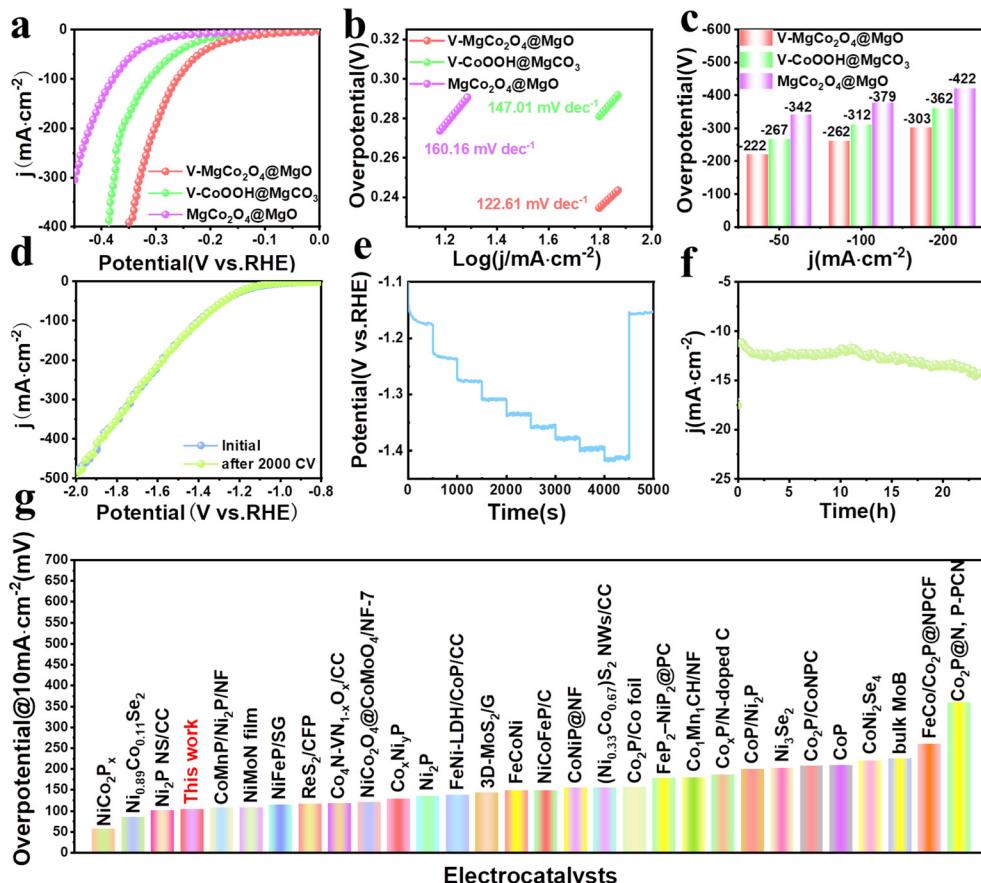


Fig. 4 Evaluation of HER performance. (a) LSV curves and (b) Tafel plots of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$, $\text{V-CoOOH}@{\text{MgCO}_3}$ and $\text{MgCo}_2\text{O}_4@\text{MgO}$ at a scan rate of 5 mV s^{-1} . (c) The required overpotentials for $\text{V-MgCo}_2\text{O}_4@\text{MgO}$, $\text{V-CoOOH}@{\text{MgCO}_3}$ and $\text{MgCo}_2\text{O}_4@\text{MgO}$ to drive -50 , -100 , and -200 mA cm^{-2} . (d) LSV curves of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ before and after 2000 CV cycles. (e) Multicurrent process obtained with the $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ electrode in 1.0 M KOH without IR correction. The current density ranges from 10 mA cm^{-2} to 100 mA cm^{-2} , with an increment of 10 mA cm^{-2} per 500 s . (f) Time-dependent current density curve for $\text{V-MgCo}_2\text{O}_4@\text{MgO}$. (g) Comparison of overpotential for $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ at 10 mA cm^{-2} with other electrocatalysts recently reported in 1 M KOH .

using an inductively coupled plasma optical emission spectrometer (ICP-OES). The content of Mg, Co and O elements in $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ was 13.7 , 53.9 and 31.1 wt\% , respectively. As the precursor to prepare $\text{V-MgCo}_2\text{O}_4@\text{MgO}$, $\text{V-CoOOH}@{\text{MgCO}_3}$ (Fig. 2b) presents XRD peaks of CoOOH (JCPDS No. 73-1213) and MgCO_3 (JCPDS No. 08-0479). Fig. 2d shows the X-ray photoelectron spectroscopy (XPS) survey spectrum of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$, evidencing the existence of Mg, Co, O and V elements, consistent with EDX results (Fig. 1j). The measured O 1s XPS spectrum of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ presents three peaks at ~ 532.8 , ~ 531 and $\sim 529 \text{ eV}$ (Fig. 2e). The 532.8 eV peak is related to water on the $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ surface. The 531 eV signal is ascribed to the oxygen vacancy. The binding energy at 529 eV is due to the typical metal–oxygen bond.³⁹ The V 2p_{2/3} peaks involve two subpeaks at 515.7 and 516.6 eV (Fig. 2f), which may be attributed to distinct valence states of V (V^{3+} and V^{4+}).⁴⁰ The large electronegativity of high-valence-state V^{4+} in $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ can achieve strong electronic affinity synergy with Mg/Co sites. For the $\text{MgCo}_2\text{O}_4@\text{MgO}$ sample, the V 2p XPS signal is absent (Fig. S3†). The characteristic XPS peak of Mg

1s for $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ is observed at 1303.5 eV (Fig. 2g).⁴¹ Two main peaks of Co 2p at 780.6 and 796.6 eV (Fig. 2h) are assigned to Co 2p_{3/2} and Co 2p_{1/2}, respectively, suggesting the existence of Co^{3+} .⁴² Differing from $\text{MgCo}_2\text{O}_4@\text{MgO}$ (Co 2p_{3/2}: 780.38 eV ; Co 2p_{1/2}: 796.58 eV , Fig. S3†), the larger Co 2p binding energies of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ reveal the higher oxidation state of Co atoms in $\text{V-MgCo}_2\text{O}_4@\text{MgO}$.

The OER performance of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ was studied using a standard three-electrode system in 1 M KOH at room temperature. The precursor to prepare $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ (vanadium-containing $\text{CoOOH}@{\text{MgCO}_3}$, denoted as $\text{V-CoOOH}@{\text{MgCO}_3}$), and $\text{MgCo}_2\text{O}_4@\text{MgO}$ are also used as comparative samples (Fig. 3a). It can be seen that the overpotential of $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ for 100 mA cm^{-2} is about 240 mV , which is significantly lower than that of $\text{V-CoOOH}@{\text{MgCO}_3}$ ($\eta_{100} = 361 \text{ mV}$) and $\text{MgCo}_2\text{O}_4@\text{MgO}$ ($\eta_{100} = 394 \text{ mV}$), indicating that the introduction of vanadium is of great benefit for improving OER performance. The OER kinetics of the catalyst can be evaluated using the Tafel slope (Fig. 3b). The smaller Tafel slope for $\text{V-MgCo}_2\text{O}_4@\text{MgO}$ ($108.13 \text{ mV dec}^{-1}$) than $\text{V-CoOOH}@{\text{MgCO}_3}$ ($113.24 \text{ mV dec}^{-1}$)

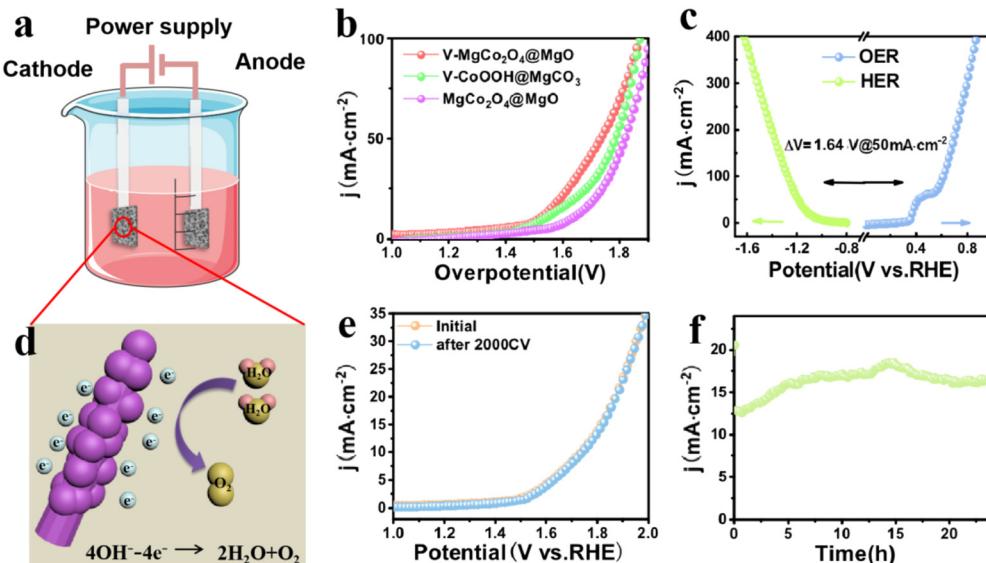


Fig. 5 Evaluation of overall water splitting performance. (a) Schematic of the two-electrode water-splitting electrolyzer with V-MgCo₂O₄@MgO as both the anode and cathode. (b) LSV curves of V-MgCo₂O₄@MgO|V-MgCo₂O₄@MgO, MgCo₂O₄@MgO|MgCo₂O₄@MgO, and V-CoOOH@MgCO₃ two-electrode configurations at a scan rate of 5 mV s⁻¹. (c) The polarization curves of V-MgCo₂O₄@MgO for the HER and OER. (d) Schematic diagram of V-MgCo₂O₄@MgO for water splitting performance in 1 M KOH. (e) LSV curves of V-MgCo₂O₄@MgO|V-MgCo₂O₄@MgO before and after 2000 CV cycles. (f) Time-dependent current density curve for the two-electrode configuration of V-MgCo₂O₄@MgO|V-MgCo₂O₄@MgO.

and MgCo₂O₄@MgO (134.5 mV dec⁻¹) reveals the faster OER kinetics and higher charge transfer rate of V-MgCo₂O₄@MgO. V-MgCo₂O₄@MgO only needs overpotentials of 261.5 and 274.7 mV for 150 and 200 mA cm⁻² (Fig. 3c), respectively, significantly lower than V-CoOOH@MgCO₃ ($\eta_{150} = 379.5$ mV; $\eta_{200} = 393$ mV) and MgCo₂O₄@MgO ($\eta_{150} = 419.2$ mV; $\eta_{200} = 437$ mV). Cyclic voltammetry (CV) test was conducted to measure the double-layer capacitance (C_{dl}) with scan rates ranging from 10 to 100 mV s⁻¹ (Fig. S4a-c†). Based on C_{dl} values (Fig. 3d), the electrochemical surface area (ECSA) can be calculated (Fig. S4d†). The C_{dl} (51.27 mF cm⁻²) and ECSA (320.44 cm²) of V-MgCo₂O₄@MgO were larger than those of V-CoOOH@MgCO₃ (C_{dl} : 42.08 mF cm⁻²; ECSA: 263 cm²) and MgCo₂O₄@MgO (C_{dl} : 32.66 mF cm⁻²; ECSA: 204.13 cm²), indicating the better active-site exposure of V-MgCo₂O₄@MgO. In order to comprehensively evaluate V-MgCo₂O₄@MgO, we also investigated the catalytic stability of the OER by multistep chronopotentiometry (Fig. 3e), setting the current from 10 mA cm⁻² to 100 mA cm⁻². It can be seen that the potential of V-MgCo₂O₄@MgO responds rapidly to a certain value and can be stable for 500 s. Finally, we drove a long-term chronoamperometric test for 24 h at 500 mA cm⁻² (Fig. 3f). Clearly, after 24 h, V-MgCo₂O₄@MgO still presents 400 mA cm⁻² catalytic current density (80% of the initial current), suggesting the long-term stability of V-MgCo₂O₄@MgO and its potential in practical application. V-MgCo₂O₄@MgO after 24 h stability test still maintains its original morphology with almost no nanowire breakage or shedding (Fig. S5a-f†). The OER performance of V-MgCo₂O₄@MgO is comparable to some excellent catalysts reported recently (Fig. 3g and Table S1†).

In view of the good OER performance of V-MgCo₂O₄@MgO, we also evaluate the associated HER activity. As shown in Fig. 4a, V-MgCo₂O₄@MgO requires an overpotential of 222 mV for 50 mA cm⁻², lower than that of V-CoOOH@MgCO₃ ($\eta_{50} = 267$ mV) and MgCo₂O₄@MgO ($\eta_{50} = 342$ mV). To drive large current densities of 100 and 200 mA cm⁻² (Fig. 4c), V-MgCo₂O₄@MgO demands overpotentials of 262 and 303 mV, respectively, outperforming V-CoOOH@MgCO₃ ($\eta_{100} = 312$ mV; $\eta_{200} = 362$ mV) and MgCo₂O₄@MgO ($\eta_{100} = 379$ mV; $\eta_{200} = 422$ mV). As observed in Fig. 4b, the Tafel slope of V-MgCo₂O₄@MgO (122.61 mV dec⁻¹) is smaller than that of V-CoOOH@MgCO₃ (147.01 mV dec⁻¹) and MgCo₂O₄@MgO (160.16 mV dec⁻¹), indicating the faster HER kinetics of V-MgCo₂O₄@MgO. The LSV curve for V-MgCo₂O₄@MgO after 2000 consecutive CV scanning is similar to the initial polarization curve (Fig. 4d), revealing good catalytic stability. As shown in Fig. 4e, the multi-step chronopotentiometry measurement curve of V-MgCo₂O₄@MgO shows that it has a rapid and stable response to the change of HER current density. In addition, the 24 h chronoamperometric measurement curve of V-MgCo₂O₄@MgO at 10 mA cm⁻² (Fig. 4f) showed only small fluctuations. The HER performance of V-MgCo₂O₄@MgO is better than many recently reported non-noble metal electrocatalysts (Fig. 4g and Table S2†).

After the evaluation of HER and OER catalytic activities of V-MgCo₂O₄@MgO, we assemble a V-MgCo₂O₄@MgO|V-MgCo₂O₄@MgO alkaline electrolyzer to examine its catalytic ability (Fig. 5a). As shown in Fig. 5b, at a current density of 50 mA cm⁻², the voltage of the V-MgCo₂O₄@MgO catalyst is 1.69 V, which is lower than that of V-CoOOH@MgCO₃ (1.82

V) and MgCo₂O₄@MgO (1.9 V), and very close to the potential difference (ΔV) between the OER and HER of 1.64 V (Fig. 5c). The overall water decomposition process of V-MgCo₂O₄@MgO can be described in Fig. 5d. Upon continuous water electrolysis after 2000 CV cycles for the V-MgCo₂O₄@MgO||V-MgCo₂O₄@MgO alkaline electrolyzer, the LSV curve is similar to the original curve (Fig. 5e). For the 24 h chronoamperometric measurement of the V-MgCo₂O₄@MgO||V-MgCo₂O₄@MgO alkaline electrolyzer, the recorded curve fluctuates up and down only in a small range (Fig. 5f), evidencing the excellent operation stability of the V-MgCo₂O₄@MgO||V-MgCo₂O₄@MgO electrolyzer.

4. Conclusions

In summary, target-atom vanadium was implanted into the MgCo₂O₄@MgO system to cooperate with Mg sites, optimizing the Co-site electronic structure as the V-MgCo₂O₄@MgO sample through hydrothermal coordination and calcination strategies. The introduction of vanadium reverses the Mg suppression effect on Co-site water oxidation. Low-activity Co catalytic sites of MgCo₂O₄ are reactivated for large-current (500 mA cm⁻²) alkaline water oxidation with low overpotentials of 240 and 290 mV for 100 and 500 mA cm⁻², respectively, in 1 M KOH. Operating at 500 mA cm⁻², V-MgCo₂O₄@MgO has good catalytic stability for at least 20 hours. This work not only presents excellent catalysts through atomic-level structural tailoring, but also throws light on superior electronic affinity synergy between V and Mg species toward Co-O bond optimization.

Conflicts of interest

The authors declare no competing financial interest.

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