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Room-temperature ultrafast synthesis, morphology and upconversion luminescence of $K_{0.3}Bi_{0.7}F_{2.4}$:Yb³⁺/Er³⁺ nanoparticles for temperature-sensing application†

Xiaoli Gao,^{ab} Feng Song, [®]** Dandan Ju,^c Aihua Zhou, [®] Adnan Khan,^{ab} Ziyu Chen,^{ab} Xu Sang,^{ab} Ming Feng^{ab} and Lisa Liu^{ab}

An ultrafast route at room temperature has been developed for the first time to synthesize lanthanide ion (Yb^{3+}/Er^{3+}) -activated $K_{0.3}Bi_{0.7}F_{2.4}$ fluorescent nanoparticles and hold the potential to be applied in luminescent temperature sensors. In addition, the crystal structure and morphology of the K_{0.3}Bi_{0.7}F_{2.4} nanoparticles prepared at different reaction conditions were studied in detail for the first time, including the reaction time and different mole ratios of Bi source and NH₄F. Moreover, Yb³⁺/Er³⁺ doped in the K_{0.3}Bi_{0.7}F_{2.4} matrix using a luminescent thermometer based on the fluorescence intensity ratio technology was also investigated. Additionally, K_{0.3}Bi_{0.7}F_{2.4}:Yb³⁺/Er³⁺ revealed excellent up-conversion luminescence performance, and the up-conversion mechanisms of samples are affirmed according to the up-conversion emission intensity as a function of the pump power density. Simultaneously, the optical thermometric performance of the samples was investigated in the temperature range of 323-523 K to indagate the temperature sensor performance. Significantly, the maximum sensor sensitivity of the studied nanoparticles were 0.0058 K⁻¹ (S_a) at 523 K and 1.08 K⁻¹ (S_r) at 323 K, which indicates that $K_{0.7}Bi_{0.7}F_{2.4}$:20%Yb³⁺/2%Er³⁺ fluorescent nanoparticles can be exploited as a promising luminescent thermometer.

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1. Introduction

Lanthanide ions (Ln3+) activated up-conversion luminescence materials have been applied in many fields including threedimensional displays, white light-emitting diodes, biological imaging, temperature sensors and photocatalysts because of their unique optical merits. 1-7 In particular, non-contact optical temperature sensing has attracted considerable attention in the past few years. Fluorescence intensity ratio (FIR) is regarded as the most promising approach for non-contact temperature detection because it is independent of the two thermally coupled levels of Ln3+ ions as a function of the temperature in the measurement process.8-10 Thereinto, the thermometric

properties of Er3+ activated up-conversion materials were largely

It is known that the low phonon energy of an ideal matrix material can reduce the rate of non-radiative relaxation and maximize the luminous efficiency. During the past few decades, Ln3+-doped fluoride as a research focus has attracted considerable attention. Because fluorides have good chemical stability and low phonon energy compared to other types of compounds, which is, therefore, often used as a matrix for effectively doping rare-earth ions. 13-15

Bismuth is known as "green metal" with $[Xe]4f^{14}5d^{10}6s^26$ p³ electronic configuration because it possesses diverse oxidation states, multi-type electronic structures, nontoxic,

investigated because of their thermally compound levels of ²H_{11/2} and ⁴S_{3/2} with proper energy separation in the range of 200-2000 cm⁻¹.⁴ It was demostrated that various Er³⁺-activated green emitting phosphors were proper FIR-based optical probers. However, the sensor sensitivity is still not high enough and it needs to be addressed despite these advantages. Thus, we need to search for new optical thermometers to overcome these weaknesses. Moreover, the strong up-conversion luminescence is conducive to improving the accuracy of the temperature sensor. 11 Nevertheless, the fluorescence intensity is influenced by the host materials. Therefore, it is necessary to choose suitable host materials.¹²

^a School of Physics, The Key Laboratory of Weak Light Nonlinear Photonics, Ministry of Education, Nankai University, Tianjin, 300071, PR China. E-mail: fsong@nankai.edu.cn

^b Collaborative Innovation Center of Extreme Optics, Shanxi University, Taiyuan,

^c School of Science, Tianjin Chengjian University, Tianjin 300384, PR China

^d Physics Department, School of Science, Tianjin University of Science & Technology, Tianjin 300457, People's Republic of China

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and harmless properties. 16-19 Hitherto, we consider that K_{0.3}Bi_{0.7}F_{2.4} may also be a suitable UC host material, which is inexpensive and thermodynamically stable at high temperatures. Until now, K_{0.3}Bi_{0.7}F_{2.4} nanoparticles, as the bismuth-based host, are rarely reported for achieving Ln3+ doping. Currently, for the K_{0.3}Bi_{0.7}F_{2.4} samples, there is only one published paper on Yb3+/Tm3+ doped K0.3Bi0.7F2.4 synthesized via a hydrothermal method as an UC luminescence material for deep-tissue optical imaging.20 Thus, there is still no report on Yb3+/Er3+ co-doped K_{0.3}Bi_{0.7}F_{2.4} matrix nanoparticles for temperature sensor applications. Moreover, Yb3+/Tm3+ doped K0.3Bi0.7F2.4 were synthesized by a hydrothermal method in this study, and we did not study the matrix synthesis in detail. In our work, the $K_{0.3}Bi_{0.7}F_{2.4}:Yb^{3+}/Er^{3+}$ fluorescent nanomaterial synthesized at room temperature and the morphology of the $K_{0.3}Bi_{0.7}F_{2.4}$ nanoparticles prepared under different reaction conditions are studied for the first time in detail, such as reaction time and different mole ratios of Bi source and NH₄F.

Herein, we report an ultrafast approach at room temperature to fabricate $x\%Yb^{3+}/2\%Er^{3+}$ activated $K_{0.3}Bi_{0.7}F_{2.4}$ luminescent nanoparticles based on a chemical precipitation method in 1 min. This method is simple, efficient and low cost without the application of high temperature and pressure environment. Besides, the crystal structure and morphology of the K_{0.3}Bi_{0.7}F_{2.4} nanoparticles prepared under different reaction conditions are studied in detail. Moreover, the excitation and emission spectra of Yb3+/Er3+ doped K0.3Bi0.7F2.4 UC nanoparticles reveal excellent luminescence performance. In addition, we also investigated the UC mechanism according to the UC emission intensity as a function of the pump power density. The temperature dependence of photoluminescence intensities for the K_{0.3}Bi_{0.7}-F_{2.4}:20%Yb³⁺, 2%Er³⁺ sample was investigated at temperatures ranging from 323 K to 523 K. Furthermore, the potential application of the K_{0.3}Bi_{0.7}F_{2.4}:20%Yb³⁺, 2%Er³⁺ sample used as a temperature sensor was studied.

2. Experimental

2.1 Materials

KF (>99.9%), NH₄F (>99.9%), Bi(NO₃)₃·5H₂O(>99%) were purchased from Tianjin Damao Chem. Ethylene glycol (EG) rear-earth reagents $Yb(NO_3)_3 \cdot 6H_2O$ Er(NO₃)₃·6H₂O (>99.99%) were supplied by HWRK Chemical Company. All the reagents were of analytical grade and used without further purification.

2.2 Preparation of K_{0.3}Bi_{0.7}F_{2.4} nanoparticles and K_{0.3}Bi_{0.7}F_{2.4}:20%Yb³⁺, 2%Er³⁺ nanoparticles

A room-temperature chemical precipitation method was used to synthesize the K_{0.3}Bi_{0.7}F_{2.4} nanoparticles. First, 0.222 g NH₄F was dissolved in a 25 mL EG solution under magnetic stirring in a 50 mL glass breaker. Then, 0.58 g KF and 0.485 g Bi(NO₃)₃·5H₂O were mixed into 10 mL of EG to form a homogeneous mixture. Whereafter, the NH₄F solution was

added to the above solution under magnetic stirring for different reaction times (1 min, 5 min, 30 min, 1 h, 3 h, and 6 h). Finally, the resulting white solid products were washed extensively three times with ethanol and dried at 80 °C for

To prepare $K_{0.3}Bi_{0.7}F_{2.4}:xYb^{3+}/2\%Er^{3+}$ nanoparticles, KF (1 mmol), $Bi(NO_3)_3 \cdot 5H_2O$ (0.98-x mmol), $Yb(NO_3)_3 \cdot 5H_2O$ (x = 0.05, 0.1, 0.15, 0.2, and 0.25 mmol) and Er(NO₃)₃·5H₂O (0.02 mmol) were dissolved in 10 mL of EG. Then, a 25 mL NH₄F solution was added to the above solution under magnetic stirring for 1 min. Finally, the products were washed extensively three times with ethanol and dried at 80 °C for 12 h.

2.3 Characterization

The crystal structure and purity of the as-synthesized $K_{0.3}Bi_{0.7}F_{2.4}$ were examined using a PANalytical X'Pert Pro diffractometer with Cu K α radiation ($\lambda = 1.5406$ Å) in the 2θ range from 10° to 90° at 40 mA and 40 kV. The morphology and composition of the products were observed on a scanning electron microscope (SEM) (JEOL JMS-7500F) attached with an energy-dispersive X-ray spectrometer (EDX). Transmission electron microscopy (TEM) was performed using a Tecnai G2 F20. The Fourier transform infrared spectrum (FT-IR) of the sample was recorded on a TENSOR II FTIR spectrometer. The photoluminescence excitation and emission spectra and decay curves of samples were recorded using a Horiba Fluorolog-3 luminescence spectrometer equipped with a continuous 980 nm laser as the excitation source. Besides, the spectrum on K_{0.3}- $Bi_{0.7}F_{2.4}:20\%Yb^{3+}/2\%Er^{3+}$ sample at the range of 323 K-523 K was recorded on a Nanolog luminescence spectrometer.

3. Results and discussion

The K_{0.3}Bi_{0.7}F_{2.4} nanoparticles were synthesized using a chemical precipitation method at room temperature, and the corresponding schematic of the preparation process is shown in Fig. 1. The simple and facile synthetic method is suited for the fast production of K_{0.3}Bi_{0.7}F_{2.4} nanoparticles, which is beneficial for practical production and applications.



Fig. 1 Schematic for the preparation process for K_{0,3}Bi_{0,7}F_{2,4} nanoparticles.

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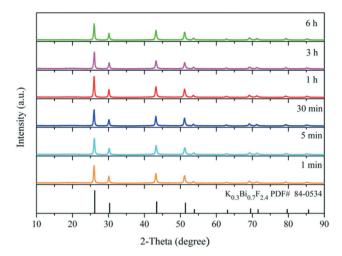


Fig. 2 XRD patterns of the $K_{0.3}Bi_{0.7}F_{2.4}$ samples prepared at different reaction times.

3.1 Phase identification and morphology of the K_{0.3}Bi_{0.7}F_{2.4} products

To gain more information about the synthesis process, a series of time-dependent experiments were carried out to evaluate the influence of the reaction time on the structure of the K_{0.3}Bi_{0.7}F_{2.4} sample with a constant concentration of NH₄F (10 mmol) in the reaction system. As revealed in Fig. 2, strong and sharp diffraction peaks can be observed for the as-prepared samples obtained at different reaction times, and all the diffraction peaks could be indexed to the standard card data of K_{0.3}Bi_{0.7}F_{2.4} (PDF#84-0534) without any trace of impurity phases, which demonstrate that the K_{0.3}Bi_{0.7}F_{2.4} crystal can be produced in a short time of one minute, and the products possess high crystallinity and phase purity. The SEM images of the K_{0.3}Bi_{0.7}F_{2.4}

nanomaterials synthesized at different reaction times were recorded to observe the morphological evolution of products. As shown in Fig. 3 and S1,† the size distribution of the asprepared nanoparticles is narrower and the average diameter is 50-60 nm. Moreover, there is almost no obvious difference in the morphology and size of the nanoparticles with the increase in the reaction time from 1 min to 1 h, even prolonging the reaction time to 3 and 6 h. The results indicate that the nucleation and growth of K_{0.3}Bi_{0.7}F_{2.4} nanoparticles can be quickly completed because the thermodynamic effect can reduce the huge surface energy of nanoparticles to achieve equilibrium of the whole system energy.21

The XRD patterns plotted in Fig. 4 clearly show pure phase K_{0.3}Bi_{0.7}F_{2.4} for all the products synthesized at different mole ratios of Bi source and NH₄F. In addition, when the mole ratio of the Bi source and NH₄F is 1:1, the yield of the precipitate in the reaction system significantly decreases, implying that the synthesis of the K_{0.3}Bi_{0.7}F_{2.4} is highly dependent on NH₄F.

In order to verify the aforementioned conjecture, a contrast experiment without using NH₄F in the reaction solution was performed. We observed that precipitate is not generated in the solution after reacting for 6 h. The results demonstrate that NH₄F plays a crucial role in the synthesis of K_{0.3}Bi_{0.7}F_{2.4} nanoparticles and the phase purity of K_{0.3}Bi_{0.7}F_{2.4} was not affected with the increase in the NH₄F content.

Fig. 5 reveals the morphological evolution of K_{0.3}Bi_{0.7}F_{2.4} samples prepared by regulating the Bi: NH₄F molar ratio. It is found from Fig. 5a that K_{0.3}Bi_{0.7}F_{2.4} samples obtained at a lower amount of NH₄F (corresponding Bi:NH₄F molar ratio of 1:1) are composed of smaller nanoparticles and the shape of the product presents quasispheres with a diameter of about 103 nm (Fig. S2a†), which is attributed to less nucleation due to the low concentration of NH₄F in the

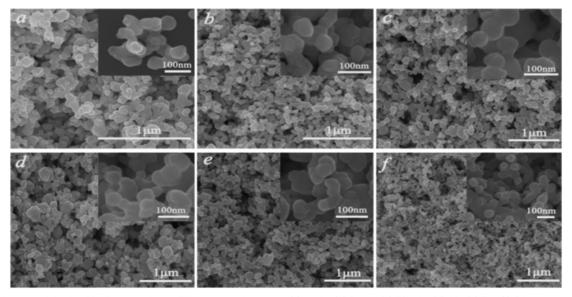


Fig. 3 SEM images of the K_{0.3}Bi_{0.7}F_{2.4} samples at different reaction times. (a) 1 min, (b) 5 min, (c) 30 min, (d) 1 h, (e) 3 h, (f) 6 h.

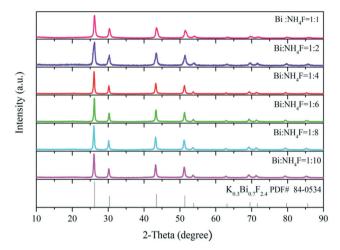


Fig. 4 XRD patterns of the K_{0,3}Bi_{0,7}F_{2,4} samples synthesized at a reaction time of 30 min with the different mole ratios of Bi source and NH₄F.

reaction system. At the Bi: NH₄F molar ratio of 1:2, the scale of spherical-like nanoparticles reduces to about 85 nm (Fig. S2b†). For the samples prepared at a Bi: NH₄F molar ratio of 1:4, the diameter of the spheroidal nanoparticles is approximately 76 nm (Fig. S2c†). With further addition of NH₄F to the synthesis system (Fig. 5d), the size of the nanoparticles was about 66 nm (Fig. S2d†). When the Bi: NH₄F molar ratios are 1:8 (Fig. 5e) and 1:10 (Fig. 5f), the mean diameter of K_{0.3}Bi_{0.7}F_{2.4} decreases to 61 nm (Fig. S2e†) and 43 nm (Fig. S2f†), respectively. According to the experimental results, the change in the size of K_{0.3}Bi_{0.7}F_{2.4} presents a downward trend with the increase in the mole

concentration of NH₄F in the synthetic system due to the nucleation number induced by the concentration of NH₄F. The results suggest that the presence of NH₄F facilitates the preparation $K_{0.3}Bi_{0.7}F_{2.4}$ samples with a smaller size.

It is well-known that fluoride as a fluorescent host can diminish the quenching of activator ions to generate effective radiative transition, particularly for UC luminescence. Thus, the Yb3+/Er3+ activated K0.3Bi0.7F2.4 nanoparticles with UC luminescence were synthesized.

The XRD patterns of as-synthesized $K_{0.3}Bi_{0.7}F_{2.4}:xYb^{3+}$, 2%Er3+ samples are illustrated in Fig. S3a.† It can be seen that all diffraction peaks of K_{0.3}Bi_{0.7}F_{2.4}:xYb³⁺, 2%Er³⁺ crystals at different concentrations of Yb3+ correspond well with the pure phase K_{0.3}Bi_{0.7}F_{2.4} (PDF#: 84-0534). Fig. S3b† shows the photoluminescence emission spectra of K_{0.3}Bi_{0.7}F_{2.4}:xYb³⁺, 2%Er³⁺ synthesized with different Yb³⁺ concentration and 2% Er3+. We note that the emission intensity of the samples increases at first with the increase in the Yb3+ concentration and then decreases when Yb3+ concentration reaches 20%. Hence, K_{0.3}Bi_{0.7}F_{2.4}:20%Yb³⁺, 2%Er³⁺ was selected in the subsequent experiments to study its luminescence characteristics and temperature sensing performance.

3.2 Phase identification, morphology and luminescence properties of the K_{0.3}Bi_{0.7}F_{2.4}:20% Yb³⁺, 2%Er³⁺ products

The structure, morphology, and UC luminescence properties of K_{0.3}Bi_{0.7}F_{2.4} nanoparticles were investigated using Er3+ as an activator and Yb3+ as a sensitizer. $K_{0.3}Bi_{0.7}F_{2.4}$:20%Yb³⁺, 2%Er³⁺ samples were

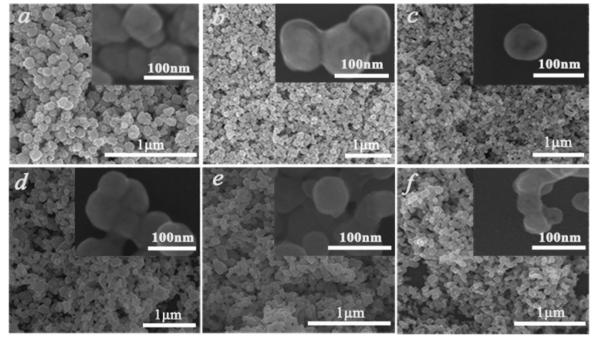


Fig. 5 SEM images of the $K_{0.3}$ Bi_{0.7}F_{2.4} samples at different mole ratios of Bi source and NH₄F. (a) Bi: NH₄F = 1:1, (b) Bi: NH₄F = 1:2, (c) Bi: NH₄F = 1:4, (d) Bi: $NH_4F = 1:6$, (e) Bi: $NH_4F = 1:8$, (f) Bi: $NH_4F = 1:10$.

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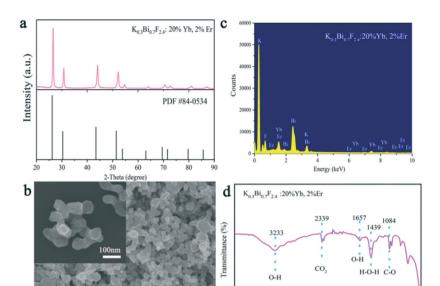


Fig. 6 XRD patterns of the $K_{0.3}Bi_{0.7}F_{2.4}$:20%Yb³⁺, 2%Er³⁺ samples (a), SEM images (b), EDX spectrum (c), and FT-IR spectrum (d) of $K_{0.3}Bi_{0.7}F_{2.4}$:20%Yb³⁺, 2%Er³⁺ samples prepared at 1 min.

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prepared under 1 min reaction time at room temperature using the chemical precipitation method. As exhibited in Fig. 6a, all the diffraction peaks of the samples obtained are well-defined with the crystal structure of K_{0.3}Bi_{0.7}F_{2.4} (PDF#: 84-0534) with lattice parameters of a = b = c = 5.89Å, $\alpha = \beta = \gamma = 90^{\circ}$, V = 204.34 Å³, and a space group of $Fm\bar{3}m(225)$. Fig. 6b reveals the SEM images of as-prepared $K_{0.3}Bi_{0.7}F_{2.4}:20\%Yb^{3+}$, $2\%Er^{3+}$ samples in 1 min. With the doping of Yb3+ and Er3+ ions, the morphology of products is consistent with that of pure K_{0.3}B_{i0.7}F_{2.4} nanoparticles and maintains a relatively narrow size distribution. The results indicate that the introduction of Yb³⁺ and Er³⁺ as doped ions did not have a significant influence on the structure and morphology of the K_{0.3}Bi_{0.7}F_{2.4} nanoparticles. To determine the element composition, the EDX spectrum (Fig. 6c) of $K_{0.3}Bi_{0.7}F_{2.4}$:20%Yb³⁺, 2%Er³⁺ nanoparticles was recorded. It can be seen that the chemical compositions of the as-synthesized samples include K, Bi, F, Yb and Er elements, and no impurity elements existed, suggesting that the Yb3+ and Er3+ ions are successfully doped into the K_{0.3}Bi_{0.7}F_{2.4} host lattice. Furthermore, the FT-IR spectrum of the $K_{0.3}Bi_{0.7}F_{2.4}$:20%Yb³⁺, 2%Er³⁺ nanoparticles was recorded to confirm the existence of EG on the surface of nanoparticles. As displayed in Fig. 6d, the peaks located at 3233 cm⁻¹ and 1657 cm⁻¹ correspond to the stretching vibrations of the O-H. The sharp peak at 1439 cm⁻¹ and 1084 cm⁻¹ belongs to the H-O-H stretching vibration and C-O stretching mode, respectively. In addition, the characteristic absorption peaks at 2339 cm⁻¹ can be ascribed to the CO2 derived from the surrounding environment.

Fig. 7a reveals the TEM image of the sample, which further verifies that the $K_{0.3}Bi_{0.7}F_{2.4}$:20%Yb³⁺, 2%Er³⁺ nanoparticles possess a spheroidal morphology. The high-resolution TEM (HRTEM) image of a selected area for $K_{0.3}$ -Bi_{0.7}F_{2.4}:20%Yb³⁺, 2%Er³⁺ nanoparticles presents a highly crystalline nature, which is consistent with the XRD results, as illustrated in Fig. 7b. The clear lattice fringes with an interplanar distance of 0.29 nm can be detected, which is in good agreement with the (200) plane of $K_{0.3}Bi_{0.7}F_{2.4}$.

Based on the HRTEM image of nanoparticles, a fast Fourier transform (FFT) pattern was obtained, which

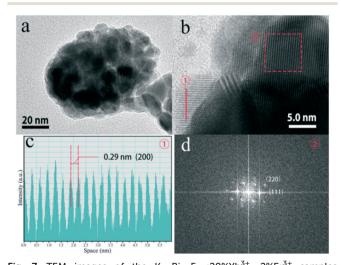


Fig. 7 TEM images of the $K_{0.3}$ Bi_{0.7}F_{2.4}:20%Yb³⁺, 2%Er³⁺ samples obtained at 1 min (a), HRTEM image (b), intensity plot of interlayer distance for the (200) plane (c) and FFT pattern (d).

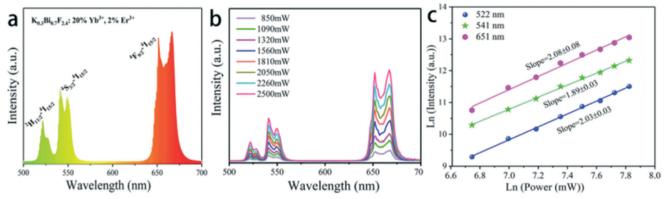


Fig. 8 Photoluminescence emission spectra of (a) $K_{0.3}Bi_{0.7}F_{2.4}$:20% Yb^{3+} , 2% Er^{3+} under 980 nm excitation, (b) UC emission spectra of the K_{0.3}Bi_{0.7}F_{2.4}:20% Yb³⁺, 2% Er³⁺ at the different pumping powers under 980 nm excitation, (c) Ln–Ln plots of the up-conversion emission intensity versus pump power for the K_{0.3}Bi_{0.7}F_{2.4}:20%Yb³⁺, 2%Er³⁺ nanoparticles under 980 nm laser radiation.

indicated a single-crystalline nature of the products and exhibited a well-defined structure with lattice fringe spacings of 0.21 and 0.34 nm corresponding to the (220) and (111) planes of $K_{0.3}Bi_{0.7}F_{2.4}$.

Fig. 8a shows the UC emission spectra of 20%Yb³⁺/2%Er³⁺ doped K_{0.3}Bi_{0.7}F_{2.4} nanoparticles. Under 980 nm NIR excitation, the spectrum of $K_{0.3}Bi_{0.7}F_{2.4}:20\%Yb^{3+}/2\%Er^{3+}$ nanoparticles exhibits the typical UC emissions of Er3+ ions. The result is consistent with previous literature.²⁰ Hereinto, the green UC emissions located at 522 and 541 nm originate from the transitions ${}^{2}H_{11/2} \rightarrow {}^{4}I_{15/2}$ and ${}^{4}S_{3/2} \rightarrow {}^{4}I_{15/2}$, respectively, and the red UC emission located at 651 nm corresponds to the ${}^4F_{9/2} \rightarrow {}^4I_{15/2}$ transition. To study the UC mechanism, we measured the UC luminescence of the samples under a series of excitation pump power (Fig. 8b). It is well known that the relationship between the UC emission intensity (I_{em}) of the samples and excitation pump power intensity (I_P) of the laser is expressed via the following formula: $I_{\rm em} \propto I_{\rm P}^{n}$, where *n* is the number of pump photons absorbed in the UC process and can be calculated from the

slope of the natural logarithmic plot. 22,23 The experimental data in Fig. 8c are fitted by the function $Ln(I_{em})$ and $Ln(I_{P})$, revealing that the slopes for ${}^{2}H_{11/2} - {}^{4}I_{15/2}$, ${}^{2}S_{3/2} - {}^{4}I_{15/2}$ and ${}^{4}F_{9/2}$ \rightarrow $^{4}I_{15/2}$ (Er³⁺, 522 nm, 541 nm, 651 nm) are 2.03 ± 0.03, 1.89 \pm 0.03 and 2.08 \pm 0.08 corresponding to two-photon (green and red) energy transfer processes. These results are consistent with previous reports. 24,25

To clearly explain the UC mechanism, the detailed schematic energy level diagram and possible UC mechanism are shown in Fig. 9. Under the 980 nm laser diode excitation, the ${}^4I_{15/2}$ level of Er^{3+} ions is excited to ${}^4I_{11/2}$ and ${}^4F_{7/2}$ levels by absorbing the energy from Yb3+ ions. Then, the 4F7/2 level nonradiatively relaxes to ²H_{11/2} and ⁴S_{3/2} states. Meanwhile, the ²H_{11/2} and ⁴S_{3/2} level can relax to the ground emitting green light near 522 nm and 541 nm.

Subsequently, the ⁴I_{11/2} state of Er³⁺ can also relax to ⁴I_{13/2} and then populate to the 4F_{9/2} energetic state by absorbing energy from Yb3+. Finally, the Er3+ in the 4F9/2 state transfers to the 4I15/2 state resulting in red light emission near 651

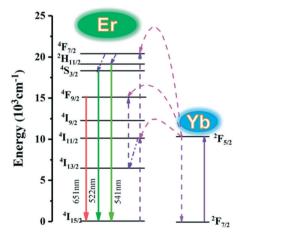


Fig. 9 Energy level diagrams showing the UC the of $K_{0.3}Bi_{0.7}F_{2.4}$:20%Yb, 2%Er upon excitation at 980 nm.

3.3 Temperature sensing of samples

Aiming at exploring the probability of the thermometry application, the temperature sensing performance of the K_{0.3}- $Bi_{0.7}F_{2.4}:20\%Yb^{3+}$ $2\% Er^{3+}$ nanoparticles investigated. It is known that the energy gap of the Er3+ between $^2H_{11/2}$ and $^4S_{3/2}$ states is about 750–850c m $^{-1}.^{28}$ Thus, the ²H_{11/2} states can be populated from ⁴S_{3/2} states because of the thermal coupling arising from this small energy separation. Finally, the emission intensity of ${}^{2}H_{11/2} - {}^{4}I_{15/2}$ and ${}^{2}S_{3/2} - {}^{4}I_{15/2}$ had variations under an elevated temperature. This characteristic indicates that K_{0.3}Bi_{0.7}-F_{2.4}:20%Yb³⁺, 2%Er³⁺ may be applied in the temperature sensor using the FIR technique. Fig. 10a shows the normalized green emission spectra of Er³⁺ ions by 980 nm excitation (laser power = 1.5 W) at the temperature range from 323 to 523 K. As plotted in Fig. 10a, it is clear that the emission intensity ratio of the ${}^{2}H_{11/2} - {}^{4}I_{15/2} / {}^{2}S_{3/2} - {}^{4}I_{15/2}$

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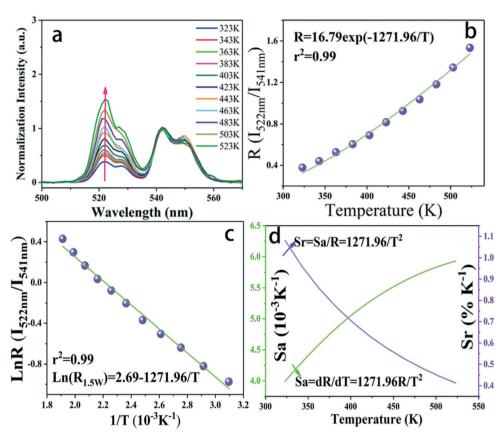


Fig. 10 (a) Temperature-dependent normalized green emission spectra of $K_{0.3}Bi_{0.7}F_{2.4}$:20%Yb³⁺, 2%Er³⁺ nanoparticles under 980 nm excitation (laser power = 1.5 W). (b) Relationship between R and the absolute temperature. (c) Monolog plots of Ln(R) as a function of inverse absolute temperature. (d) The sensitivity of $K_{0.3}Bi_{0.7}F_{2.4}$:20%Yb³⁺, 2%Er³⁺ nanoparticle as a function of absolute temperature.

increases with the increase in temperature because of the Boltzmann-type distribution. According to the Boltzmann's equation, the fluorescence intensity ratio (R) can be defined as follows:^{29,30}

$$R(I_{522\text{nm}}/I_{541\text{nm}}) = \frac{I_{522\text{nm}}}{I_{541\text{nm}}} = C \exp\left(-\frac{\Delta E}{k_{\text{B}}T}\right)$$
 (1)

In which $I_{522\mathrm{nm}}$ and $I_{541\mathrm{nm}}$ denote the relative integrated intensities corresponding to ${}^2\mathrm{H}_{11/2}{}^{-4}\mathrm{I}_{15/2}$ and ${}^2\mathrm{S}_{3/2}{}^{-4}\mathrm{I}_{15/2}$ transitions, respectively; ΔE is the energy gap between the 522 nm and 541 nm; C is the proportional constant; k_B and T are defined as the Boltzmann constant and the absolute temperature, respectively. It is noticeable that the intensity ratio $R(I_{522\mathrm{nm}}/I_{541\mathrm{nm}})$ and $\mathrm{Ln}\,R(I_{522\mathrm{nm}}/I_{541\mathrm{nm}})$ can be well-fitted with relevant equations in Fig. 10(c) and (d) with good linear fitting degree values (r^2 value is 0.99). The slope and intercept values given by the linear fitting result are 2.69 and 1271.96, respectively. At last, the ΔE and constant C are 0.11 eV and 16.79, respectively. Besides, sensitivity is also a significant parameter to evaluate the temperature sensing application. It is worth investigating, which can be determined by: ${}^{31-33}$

$$S_{\rm a} = \frac{{\rm d}R}{{\rm d}T} = R \frac{\Delta E}{k_{\rm B}T^2} \tag{2}$$

Here, the calculated sensitivity values increased upon elevating the temperature to 523 K. The maximum sensitivity is about 0.0058 K⁻¹ at 523 K. Furthermore, compared to other results on previously developed optical thermometers, the $K_{0.3}Bi_{0.7}F_{2.4}$:20%Yb³⁺, 2%Er³⁺ nanomaterial has a relatively higher absolute sensitivity at 523 K. In addition, relative sensor sensitivity (S_r) at different temperatures was also discussed. Based on previous reports, the values of S_r as a function of temperature were calculated by utilizing the following expressions:^{33–35}

Table 1 The comparison of maximum $S_{\rm a}$, $S_{\rm r}$ temperature sensing properties of Er-doped materials

Compounds	$\lambda_{\rm ex} [{\rm nm}]$	T[K]	$S_{\rm a}\left[{ m K}^{-1} ight]$	$S_{\mathrm{r}}\left[\mathrm{K}^{1}\right]$	Ref.
α-NaLuF ₄ :Yb ³⁺ /Er ³⁺	980	150-298	0.0022	_	36
CaZnOS:Er ³⁺	980	303-603	0.0033	_	37
NaBiF ₄ :Er ³⁺ /Yb ³⁺ /Fe ³⁺	980	303-543	0.0053	_	38
Bi ₄ Ti ₃ O ₁₂ :Er ³⁺	980	115-490	0.0043	_	39
$KEr_3F_{10}:Er^{3+}$	980	303-573	0.00398	_	40
BiPO ₄ :Er ³⁺ /Yb ³⁺	980	313-573	0.00413	1.10%	41
BiOCl:Yb ³⁺ , Er ³⁺	980	298-778	0.0028	_	42
Bi ₃ Ti _{1.5} W _{0.5} O ₉ :Er ³⁺	980	83-423	0.00314	_	43
Na _{0.5} Bi _{0.5} TiO ₃ :Er/Yb-	980	93-613	0.0031	_	44
Er, Yb:GdVO ₄ @SiO ₂	980	297-343	_	1.01%	45
$K_{0.3}Bi_{0.7}F_{2.4}$: Yb^{3+}/Er^{3+}	980	323-523	0.0058	1.08%	This work

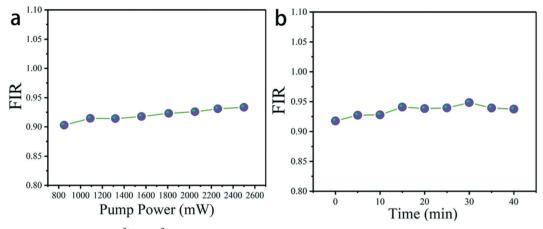


Fig. 11 (a) FIR of the K_{0.3}Bi_{0.7}F_{2.4}:20%Yb³⁺, 2%Er³⁺ nanoparticles under 980 nm excitation at different power at room temperature, (b) FIR of the $K_{0.3}Bi_{0.7}F_{2.4}:20\%Yb^{3+}$, $2\%Er^{3+}$ nanoparticles under 1.5 W (980 nm) excitation at persistent irradiation measured within 40 min.

$$S_{\rm r} = \frac{1}{R} \frac{\mathrm{d}R}{\mathrm{d}T} = \frac{\Delta E}{k_{\rm B}T^2} \tag{3}$$

As depicted in Fig. 10d, the corresponding resultant value of S_r as a function of temperature has been given. In the detection range of 323-523 K, the value of S_r monotonously decreases with temperature increases and the maximum value was about 1.08% K⁻¹ at the temperature of 323 K. Hence, the consequence implies that the sample may have potential applications in temperature sensing using the FIR technique. The temperature sensing properties of other Erdoped materials, which include fluorides, phosphate, vanadate and tungstate, are listed in Table 1.

up-conversion power and time-dependent luminescence spectra of $K_{0.3}Bi_{0.7}F_{2.4}:20\%Yb^{3+}$, $2\%Er^{3+}$ are plotted in Fig. 11a to verify the heating effect caused by 980 laser excitation. In addition, the luminescence spectra were recorded under 1.5 W (980 nm) excitation at persistent irradiation measured within 40 min in Fig. 11b. 45,46 As it can be seen that for the different powers of 980 nm excitation the value of FIR barely changed. Therefore, no pump induced heating is observed.46 Furthermore, the value of FIR remained almost constant under persistent irradiation measured within 40 min, manifesting that the temperature of the material does not change greatly.

4. Conclusions

In summary, we have successfully developed an ultrafast synthetic strategy to prepare K_{0.3}Bi_{0.7}F_{2.4} nanoparticles based on the facile chemical precipitation method at room temperature. The shape of the product is the spherical-like nanoparticles, and the average diameter of the samples decreased by regulating the mole ratio of the Bi source and NH₄F. However, the average diameter of the samples is no longer changed based on different reaction times. Furthermore, we have investigated the UC Photoluminescence properties and UC mechanism. In addition, this new K_{0.3}Bi_{0.7}F_{2.4}:20%Yb³⁺, 2%Er³⁺ phosphors could be used as a potentially useful temperature sensor utilizing the FIR technology due to the temperature-dependent property of K_{0.3}Bi_{0.7}F_{2.4}:20%Yb³⁺, 2%Er³⁺ phosphors revealing a linear relation at 323-523 K under 980 nm excitation and relatively high sensitivity.

Conflicts of interest

There are no conflicts to declare.

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