



Nonlinear ion transport in a 2D Janus membrane with an angstrom pore: memristive and negative differential resistance phenomena

Mohan Teja Dronadula ^{ab} and Narayana R. Aluru ^{*ab}

Received 28th November 2025, Accepted 21st January 2026

DOI: 10.1039/d5fd00130g

We report nanofluidic memristive behavior and negative differential resistance (NDR) in a MoSSe angstrom pore using molecular dynamics (MD) simulations. The system exhibits pinched-loop hysteresis in its current–electric field (I – E) response when subjected to a sinusoidal electric field pulse. Furthermore, we identified a NDR region, marked by a conductance switch from low to high resistance beyond a critical threshold electric field. This switching behavior originates from electric field induced dipole ordering of water confined within the nanopore, leading to an increased translocation barrier. We further show that the system exhibits biological synapse like behavior, exhibiting neuromorphic functions such as short-term plasticity and paired-pulse facilitation/depression. These findings demonstrate that nanoscale confined solvent ordering can serve as a physical basis for achieving non-linear ion transport behavior, enabling applications in next-generation neuromorphic computing systems.

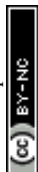
1. Introduction

Synapses are specialized connections between neurons in biological neural networks that enable the transmission, modulation and storage of information. These connections are inherently plastic, a property that underlies the brain's capacity for learning and adaptation.¹ Synapses achieve this by adjusting their efficacy (which acts as an internal state variable) based on the history of neural activity, thus embodying past inputs as memory. Through this dynamic, activity-dependent plasticity, synapses simultaneously encode prior activity and process new signals, serving as the fundamental substrate for both information storage and computation in the brain.

Memristors have emerged as novel electronic components capable of emulating this coupled memory–computation behavior of biological synapses. Their appeal lies in unifying processing and storage within a single element,

^aWalker Department of Mechanical Engineering, The University of Texas at Austin, Austin, Texas 78712, USA

^bOden Institute for Computational Engineering and Sciences, The University of Texas at Austin, Austin, Texas 78712, USA. E-mail: aluru@utexas.edu; Web: <https://sites.utexas.edu/aluru/>



thereby overcoming the von Neumann bottleneck.² The concept was first theorized by Leon Chua in 1971³ as the missing fourth fundamental circuit element that relates electric charge with magnetic flux. It was later realized experimentally⁴ in 2008. Since then, numerous studies have investigated solid-state memristors based on modulating electron/hole transport using various phenomena.^{4–8} More recently, attention has shifted toward nanofluidic memristors, with several studies demonstrating their operation through a range of nanoscale phenomena, including ion induced distortion of solvent ordering or networking,⁹ electro-wetting,¹⁰ concentration polarization,^{11,12} ion adsorption/desorption,^{13,14} the Wien effect,^{15,16} electrostriction,¹⁷ *etc.* These systems function analogously to their electronic counterparts but utilize ions rather than electrons, allowing operation at lower voltages and reduced power consumption.^{13,18} When driven by signals on timescales comparable to the relaxation timescales of the underlying processes, they exhibit pinched hysteresis loops (a hallmark of memristors) and activity-dependent plasticity reminiscent of biological neurons.

Beyond such memristive hysteresis, another important nonlinear ion transport behavior is negative differential resistance (NDR), in which an increase in applied voltage produces a decrease in current over a finite bias range. This characteristic plays a crucial role in circuit implementations of the FitzHugh–Nagumo (FN) neuron model,^{19,20} where it provides the nonlinear element required for generating excitability and spike-like dynamics observed in neurons. Numerous solid state NDR memristors have been demonstrated and successfully employed to emulate such spiking behavior,^{20–24} whereas only a few nanofluidic systems have exhibited such capabilities.^{14,25,26}

In this study, we present a nanofluidic NDR memristor based on angstrom pores in a Janus MoSSe membrane. By applying a sinusoidal electric field across the membrane, we observe pinched loop hysteresis together with NDR. We investigate the mechanism underlying these behaviors and find that the electric field induces a pronounced alignment of water dipoles inside the nanopore. We show that the dipole orientation strengthens with electric field, and the translocation barrier for ions correspondingly increases, leading to a reduction in current and the emergence of NDR. We also explain that in the reverse sweep, due to the longer timescale associated with the relaxation of the dipole-oriented water structure compared to the forward process, the system retains a more ordered state, resulting in lower currents during the reverse sweep compared to the forward sweep. We further demonstrate biological neuron-like short-term plasticity, exhibiting both short-term facilitation (STF) and short-term depression (STD). Paired-pulse facilitation (PPF) and paired-pulse depression (PPD) are also observed, allowing us to extract the characteristic memory timescales exhibited by our system. Additionally, current–voltage curves obtained for different membrane charge states show that the water ordering mediated mechanism persists even in uncharged systems. Overall, this study reveals a unique confined water ordering induced nonlinear ion transport, with promising implications for next generation iontronic neuromorphic computing devices.

2. Methods

The simulated system consists of two water reservoirs containing 1 M KCl, placed on either side of a $5.2 \times 5.07 \text{ nm}^2$ MoSSe membrane, as shown in Fig. 1A. The



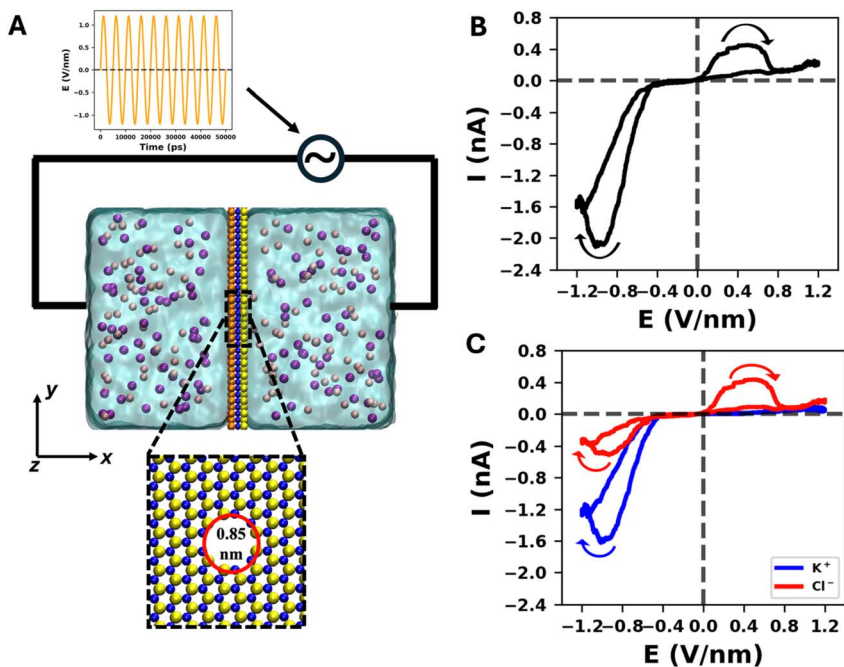


Fig. 1 (A) Simulation set-up showing system, membrane, nanopore and electric field signal used. Here, Se, Mo and S atoms are represented by orange, blue and yellow spheres, respectively. (B) Current–electric field (I – E) characteristics. The arrow indicates the loop direction. (C) Decomposition of the total current in (B) into K^+ and Cl^- contributions.

membrane contains a single nanopore with a center-to-center diameter of 0.85 nm (Fig. 1A). After equilibrating the system for 10 ns in NVT ensemble, we applied a series of ten sinusoidal electric field pulses, as illustrated in Fig. 1A. Each pulse had an amplitude $A = 1.2 \text{ V nm}^{-1}$ and a frequency $f = 200 \text{ MHz}$. All simulations were performed using the GROMACS^{27–30} simulation package with a timestep of 1 fs at 300 K, employing a Nosé–Hoover^{31,32} thermostat with a relaxation time of $\tau_r = 0.1 \text{ ps}$. Short-range non-bonded interactions (Lennard–Jones and real-space Coulomb) were truncated at 1.2 nm, while long-range electrostatics were treated using the particle-mesh Ewald (PME) method.³³ Water molecules were modelled using the SPC/E water model,³⁴ and ion–ion interactions followed the Joung–Cheatham parameters.³⁵ Parameters for Mo and S were adopted from previous work,³⁶ while Se parameters were taken from the CHARMM forcefield.³⁷ Cross interactions were derived using the Lorentz–Berthelot mixing rules. The atomic charges assigned to Mo, S, and Se in the MoS₂ membrane were $+1.06e$, $-0.61e$, and $-0.45e$, respectively. For MoS₂, the charges were $+1.18e$ for Mo and $-0.59e$ for S. These values were obtained using the Bader charge partitioning scheme.³⁸ The MoS₂ charges closely match the values reported in a previous work,³⁹ while the MoS₂ charges are consistent with those reported elsewhere.⁴⁰ It should be noted that the precise magnitude of atomic charges does not play a crucial role in NDR or hysteresis (discussed in the Results section) as both mechanisms persist, albeit to a varying degree depending on the charges used. Unless otherwise noted, all plots were obtained by averaging over 10



independent ensembles, except the PPF/PPD and STF/STD results, which were averaged over 150 ensembles.

3. Results

3.1 Origin of negative differential resistance

The current–electric field (I – E) response shown in Fig. 1B exhibits NDR beyond $E = 0.5 \text{ V nm}^{-1}$ for positive electric fields and beyond $E = -1.0 \text{ V nm}^{-1}$ for negative electric fields. Furthermore, the I – E curve displays a pinched-loop hysteresis with unipolar memory features, where the reverse sweep currents are lower compared to forward sweep currents. Decomposition of the current into individual ionic (K^+ and Cl^-) contributions (Fig. 1C) revealed that both ions exhibit similar features of NDR and hysteresis. To understand the mechanistic origin of the NDR phenomenon, we investigated the Cl^- ion translocation energy barriers ($\Delta\omega$) as a function of electric field. These energy barriers are potential of mean force (PMF) barriers estimated as $\Delta\omega = -\int_{r_1}^{r_2} \langle F(x) \rangle dx$, where “ $\langle F(x) \rangle$ ” denotes the mean force, and r_1 and r_2 are the positions of the energy minima and maxima, respectively. Table 1 reports these energy barriers and their decompositions for different electric fields in the NDR region.

The total energy barriers show a decrease from 0.2 V nm^{-1} to 0.4 V nm^{-1} , and an increase from the 0.4 V nm^{-1} to 0.8 V nm^{-1} aligning with the NDR phenomenon. Further, the decompositions show that this increase in energy barriers causing NDR is primarily due to ion–water interactions, more specifically, electrostatic interactions. A similar phenomenon was previously reported for hydrophobic nanopores by Hansen *et al.*,⁴¹ where the application of an electric field increased the ion translocation energy barrier. This was explained to be caused by the electric field induced alignment of water dipoles producing an induced electric field that opposed ion motion. Upon inspecting the water structure inside the nanopore in our system, we noticed a similar phenomenon; the application of an electric field results in the water molecules inside the nanopore reorienting themselves such that their dipoles align parallel to the applied electric field, as shown in Fig. 2A. Further, we quantified the extent of this alignment by projecting the dipole moment of each water molecule inside the nanopore onto the x – y plane and computing the angle between this projection and the x –axis (Fig. 2A). We then calculated the probability distribution function of these angles and fit a Gaussian

curve $P(\theta) = \frac{1}{\sqrt{2\pi}\sigma_\theta} e^{-\frac{(\theta-\mu)^2}{2\sigma_\theta^2}}$ to the distribution, where “ μ ” and “ σ_θ ” are the mean

Table 1 Energy barriers and decompositions for Cl^- ions for different electric fields in the forward sweep. The energies are in kJ mol^{-1} . The total energy barriers also contain energy from applied external field

	Water		Membrane		Ion	Total	
	VdW	Electrostatic	VdW	Electrostatic	VdW		Electrostatic
0.2 V nm^{-1} (forward)	13.67	−21.52	1.23	−24.775	−2.85	39.91	4.05
0.4 V nm^{-1} (forward)	16.87	4.38	−1.76	−24.74	−2.19	12.745	2.03
0.6 V nm^{-1} (forward)	15.155	18.77	3.08	−23.18	−0.48	−4.8	4.8
0.8 V nm^{-1} (forward)	16.54	32.22	3.92	−21.97	−0.00035	−19.1	6.2



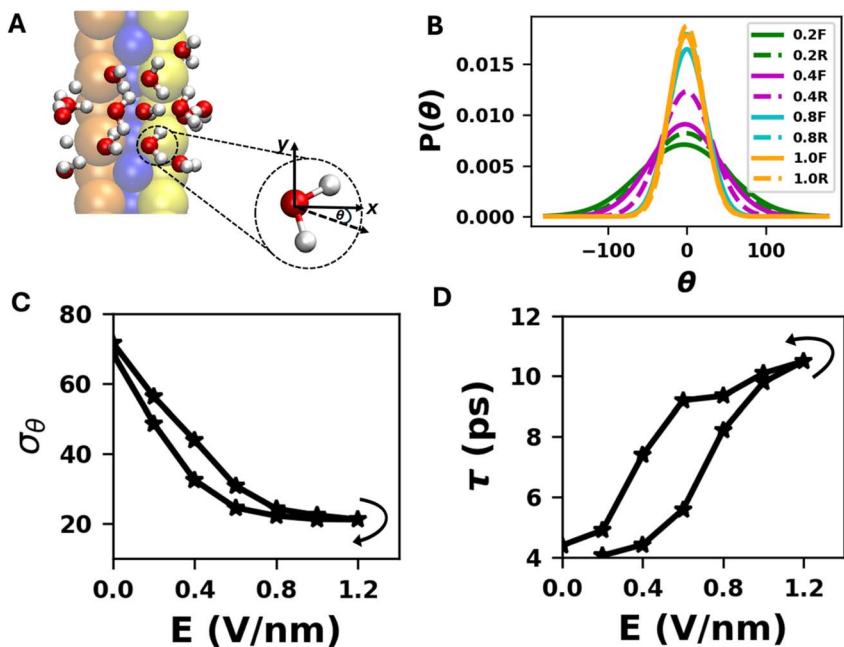


Fig. 2 (A) Snapshot showing the nanopore with water molecules oriented parallel to applied field for positive electric fields. Here, the membrane is in the y - z plane with pore axis oriented along the x -axis and the field is applied in the positive x -direction, " θ " is the angle between the projection of the dipole in the x - y plane and the x -axis. (B) Distribution of water dipole orientation plots for various electric fields in the forward and reverse sweeps. (C) Standard deviation of water dipole orientation distributions vs. electric field in the forward and reverse sweeps. (D) Water orientation relaxation times " τ " vs. electric field in the forward and reverse sweeps.

and standard deviation, respectively. Fig. 2B shows these distributions for various electric fields in the forward and reverse sweeps and the standard deviations of these distributions are shown as a function of electric field in Fig. 2C.

Together, Fig. 2B and C show that as the electric field increases during the forward sweep, the distributions become narrower (standard deviation reduces), indicating stronger orientation of the water molecules along the applied field. This stronger orientation would result in an induced electric field that acts against the applied field, thus impeding ion motion and causing the observed NDR. Furthermore, we expect the same mechanism to operate in the negative electric field regime as well. However, the I - E curves show that NDR originates at a higher applied electric field in the negative regime. This phenomenon can be explained by the presence of an intrinsic electric field inside the nanopore, due to the Janus nature of the membrane. As shown in Fig. 1, the membrane has molybdenum sandwiched between sulfur (more electronegative) on the right and selenium (less electronegative) on the left. This results in an intrinsic electric field in the nanopore directed parallel to the applied field (strengthening it) for positive biases and antiparallel to the applied field (weakening it) for negative biases. As a result of this weakening of field for negative biases, NDR is observed at a higher magnitude of applied electric field.



3.2 Origin of hysteresis

Fig. 2B and C show that the water orientation distributions are narrower (smaller standard deviation) in the reverse sweep, compared to the forward sweep. This explains the lower currents in the reverse sweep, where hysteresis in the water structure leads to a more oriented configuration compared to the forward sweep. As explained in the previous section, this oriented water structure increases the ion translocation barrier, thereby impeding ion motion and reducing flux. To further understand the origin of this hysteresis in water structure, we looked at the dynamics of this water structure orientation process and extracted the associated timescales as a function of electric field in the forward and reverse sweeps, as shown in Fig. 2D. Fig. 3 shows the forward and reverse orientation processes in time, where we start from a relaxed system and gradually increase the electric field in the forward process (Fig. 3A) and reduce the field in the reverse process (Fig. 3B). Here, " σ_θ " is the standard deviation of the water orientation distribution at each timestep. The plots show that in the forward sweep, as the applied field is increased, there is an exponential decrease in σ_θ , whereas in the reverse sweep, as the field is decreased, there is an exponential increase in σ_θ . We extracted the timescales " τ " shown in Fig. 2D, by fitting $\sigma_\theta = A + Be^{-t/\tau}$ in the forward sweep and $\sigma_\theta = A + B(1 - e^{-t/\tau})$ in the reverse sweep. These timescales show that in the forward sweep, the water orientation process happens on faster timescales compared to the relaxation process indicating that the system holds memory in the form of oriented water structure. This explains the narrower water orientation distributions in the reverse sweep compared to the forward sweep (Fig. 2B and C) and the hysteresis in the ionic current.

Overall, the mechanism of NDR and hysteresis can be described as follows – as the electric field increases, the water molecules inside the nanopore become more oriented, with their dipoles pointing along the applied field direction. This re-orientation leads to a larger Cl^- translocation energy barrier as the electric field is increased, causing the NDR. During the reverse sweep, as the electric field

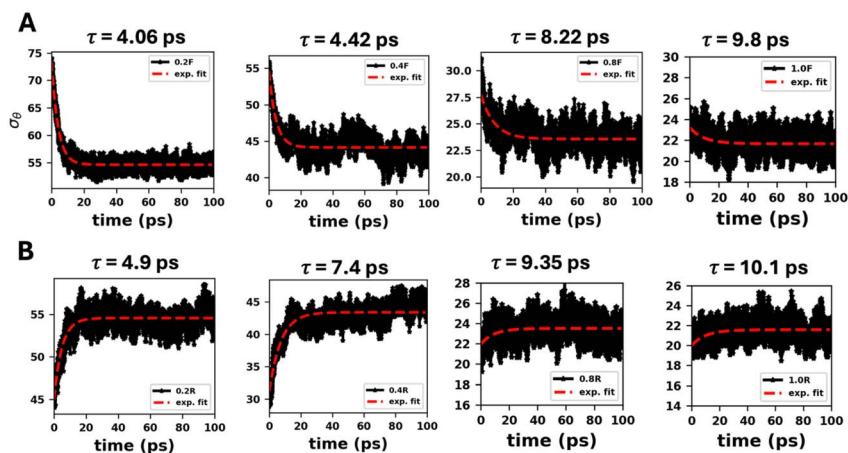


Fig. 3 Standard deviation of the water orientation distributions as a function of time for different electric fields in (A) forward and (B) reverse sweeps. Here τ is the timescale extracted from the exponential fit.



decreases, the ordered water structure relaxes on a longer timescale compared to the forward process. This delayed relaxation results in a narrower water distribution, higher Cl^- ion translocation barriers, and lower currents during the reverse sweep, leading to the observed I - E hysteresis. Furthermore, the intrinsic field of the membrane in the nanopore causes the NDR to occur at a higher applied electric field magnitude in the negative field regime. Finally, to confirm that the NDR and hysteresis originate from the water structure rather than the membrane charges, we repeated the calculations using membranes with modified S and Se charges to impose higher or lower charge asymmetry (Fig. 4). We also examined a fictitious charge neutral membrane (Fig. 4C) and MoS_2 (Fig. 4D). In all cases, NDR and hysteresis persisted, reinforcing that both phenomena arise from the water structure inside the nanopore, with the Janus nature of the membrane enhancing these phenomena.

3.3 Emulating synaptic functions

One of the key features exhibited by synapses between biological neurons that enables information processing and short-term memory is short-term plasticity (STP). This short-term synaptic plasticity is governed by a finite memory timescale of the synapse, which dictates how long the influence of a preceding signal persists. To demonstrate an analogous behavior in our system and to quantify this memory timescale, we performed paired-pulse facilitation (PPF) and paired-pulse

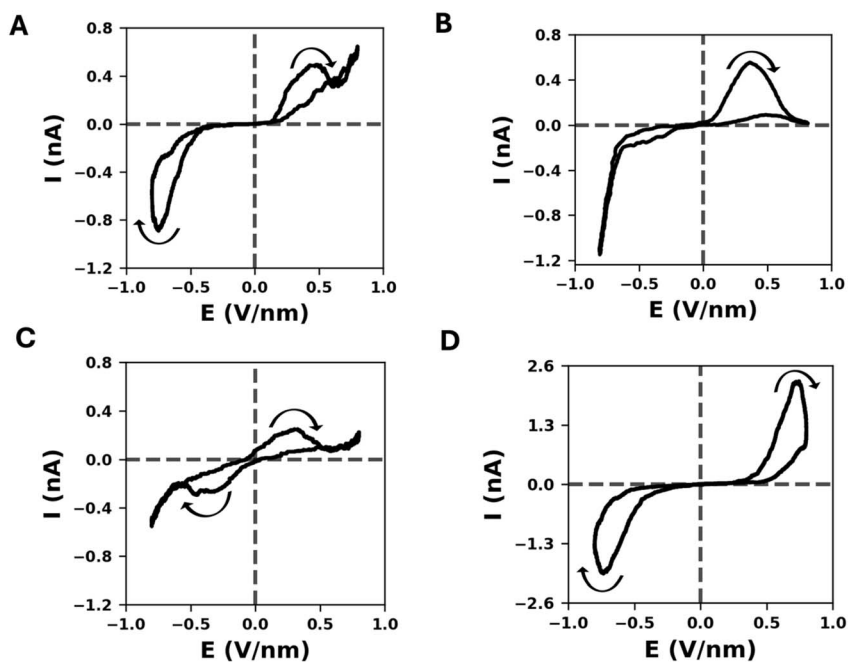


Fig. 4 I - E curves for different systems with (A) $\text{Se} = -0.53e$, $\text{Mo} = 1.06e$, $\text{S} = -0.53e$, (B) $\text{Se} = -0.35e$, $\text{Mo} = 1.06e$, $\text{S} = -0.71e$, (C) $\text{Se} = 0e$, $\text{Mo} = 0e$, $\text{S} = 0e$, as the membrane charges. (D) I - E curves for MoS_2 system with $\text{Mo} = 1.18e$, $\text{S} = -0.59e$ as the membrane charges.



depression (PPD) computations, by applying two electric field pulses with a variable delay time (Δt) in between. The ratio of the second pulse current to first pulse current (I_1/I_0) can be used to demonstrate facilitation or depression phenomena exhibited by biological synapses.

For PPF, starting from a highly oriented water configuration (generated by applying 2.0 V nm^{-1} for 20 ps), electric field pulses of 0.7 V nm^{-1} were applied with a delay period where 0.5 V nm^{-1} was applied. For PPD, starting from a relaxed system (at 0 V nm^{-1}), electric field pulses of 0.5 V nm^{-1} were applied with a delay period where no electric field was applied. Fig. 5A shows the facilitation in current which decays exponentially from $I_1/I_0 = 3.1$ to 1.2 for PPF and depression in current which grows exponentially from $I_1/I_0 = 0.2$ to 0.9 for PPD. Further, we extracted the memory timescales by fitting exponential functions $\frac{I_1}{I_0} = A_0 + A_1 \exp\left(-\frac{\Delta t}{\tau_f}\right)$ to the facilitation curve and $\frac{I_1}{I_0} = A_0 + A_1 \left(1 - \exp\left(-\frac{\Delta t}{\tau_d}\right)\right)$ to the depression curve. We extracted " $\tau_f = 8 \text{ ps}$ " and " $\tau_d = 7.5 \text{ ps}$ " as the respective memory timescales for the facilitation and depression processes in our system, indicating that the nanopore retains memory of the preceding field stimulus over a few picoseconds.

Having established $\tau = 7\text{--}8 \text{ ps}$ as the intrinsic relaxation time of our system, we investigated short-term facilitation (STF) and short-term depression (STD) using multiple pulses with a delay time between the pulses. Here, the delay time was

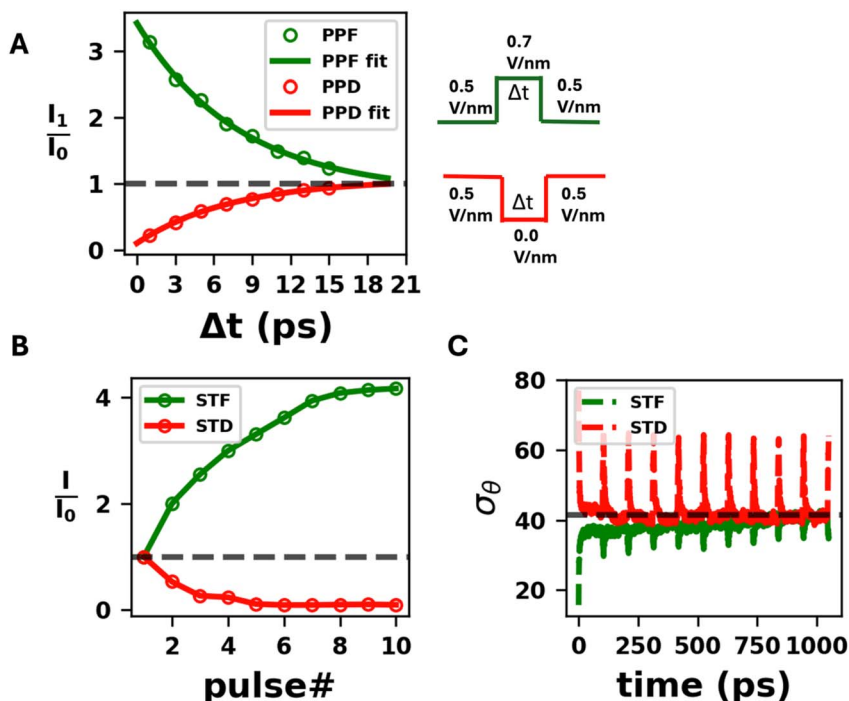


Fig. 5 (A) PPF and PPD plots along with the exponential fits, (B) STF and STD plots using $\Delta t = 5 \text{ ps}$, (C) standard deviation of the water orientation distributions as a function of time during STF/STD simulations.

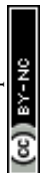


chosen as $\Delta t = 5 \text{ ps} < \tau$. In this regime, the memory of the previous pulse is not fully erased before the next one arrives, allowing cumulative plasticity to emerge. Fig. 5B shows both STF and STD behaviors exhibited by our system. For STF, we can see that the normalized current (I_n/I_0) increases over successive cycles, saturating at ~ 4 after 10 pulses. For STD, we can see the normalized current slowly decay to zero over successive pulses. Furthermore, to directly show the analogous behavior to facilitation or depression in synaptic efficacy in biological neurons, we probed the nanopore conductance as a function of time as the pulse trains are applied. Here, we used the standard deviation of the water dipole orientation distribution functions (σ_θ) as a proxy for nanopore conductance. Fig. 5C shows increase in this standard deviation (and nanopore conductance) with successive pulses during facilitation and decrease in standard deviation (and nanopore conductance) during depression.

Together, the PPF/PPD relaxation analysis and the STF/STD pulse-train results reveal a coherent picture: the sub-nanometer pore system exhibits short-term plasticity when operated in the $\Delta t < 8 \text{ ps}$ regime, where successive input signals interact through residual water orientational order, enabling both facilitation and depression analogous to biological synapses. This showcases the ability of our system to emulate key hallmarks of short-term synaptic plasticity, enabling its application in neuromorphic computing. Furthermore, while the memory timescales exhibited by our system are shorter than those typically accessible experimentally, the underlying mechanism reported here is not restricted to these timescales. The characteristic response time associated with solvent alignment depends on system parameters such as pore dimensions and properties, solvent properties, and the magnitude of the applied field. In particular, lower driving voltages, or the use of heavier polar solvents with slower dynamics⁹ are expected to shift the timescales of solvent alignment induced negative differential resistance and hysteresis to longer, experimentally accessible values.

4. Conclusions

In this work, we demonstrated that a single angstrom MoS₂ pore exhibits memristive behavior and negative differential resistance (NDR) arising from the dynamic coupling between confined water structure and ionic transport. Molecular dynamics simulations under oscillating electric fields revealed a pinched loop hysteresis in the current–electric field (I – E) characteristics and a distinct NDR region. Further analysis showed that these nonlinear transport features originate from field induced ordering of water dipoles within the nanopore, which modifies the ion–water interaction energy landscape and reduces ion mobility causing NDR. Furthermore, the delayed relaxation of this ordered water structure in the reverse sweep compared to the forward sweep leads to current hysteresis, representing the emergence of memristive behavior. The same mechanism also enables emulation of synaptic plasticity observed in biological neurons. Sequential electric field pulses produced facilitation and depression, where the nanopore conductance evolved according to the history of applied pulses. Paired pulse computations further revealed exponential relaxation kinetics, with the relaxation times (order of few picoseconds) defining the intrinsic memory timescales of the device. Together, these findings establish that solvent dipole ordering and relaxation can serve as the physical origin for



achieving non-linear ion transport behaviors like NDR and current hysteresis. More broadly, this study suggests that control of solvent order and ion–solvent coupling at the angstrom scale can enable a new class of iontronic nanofluidic memristors for neuromorphic computing applications.

Author contributions

Conceptualization: MTD, NA; methodology: MTD, NA; formal analysis: MTD; investigation: MTD; writing – original draft: MTD; writing – review & editing: NA; supervision: NA.

Conflicts of interest

There are no conflicts of interest to declare.

Data availability

All the data presented in the manuscript is available from the authors upon request.

Acknowledgements

We acknowledge the use of parallel computing resource Lonestar6 provided by the Texas Advanced Computing Center (TACC) at The University of Texas at Austin.

References

- 1 D. S. Ramon Y Cajal, Nueva Concepta De La Histologia De Los Centros Nervosos, *Ann. Surg.*, 1893, **18**, 122–124.
- 2 S. H. Jo, T. Chang, I. Ebong, B. B. Bhadviya, P. Mazumder and W. Lu, Nanoscale Memristor Device as Synapse in Neuromorphic Systems, *Nano Lett.*, 2010, **10**(4), 1297–1301, DOI: [10.1021/nl904092h](https://doi.org/10.1021/nl904092h).
- 3 L. Chua, Memristor-The Missing Circuit Element, *IEEE Trans. Circ. Theor.*, 1971, **18**(5), 507–519, DOI: [10.1109/TCT.1971.1083337](https://doi.org/10.1109/TCT.1971.1083337).
- 4 D. B. Strukov, G. S. Snider, D. R. Stewart and R. S. Williams, The Missing Memristor Found, *Nature*, 2008, **453**(7191), 80–83, DOI: [10.1038/nature06932](https://doi.org/10.1038/nature06932).
- 5 Y. V. Pershin and M. Di Ventra, Spin Memristive Systems: Spin Memory Effects in Semiconductor Spintronics, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2008, **78**(11), 113309, DOI: [10.1103/PhysRevB.78.113309](https://doi.org/10.1103/PhysRevB.78.113309).
- 6 T. Driscoll, H.-T. Kim, B.-G. Chae, M. Di Ventra and D. N. Basov, Phase-Transition Driven Memristive System, *Appl. Phys. Lett.*, 2009, **95**(4), 043503, DOI: [10.1063/1.3187531](https://doi.org/10.1063/1.3187531).
- 7 A. Chanthbouala, V. Garcia, R. O. Cherifi, K. Bouzouane, S. Fusil, X. Moya, S. Xavier, H. Yamada, C. Deranlot, N. D. Mathur, M. Bibes, A. Barthélémy and J. Grollier, A Ferroelectric Memristor, *Nat. Mater.*, 2012, **11**(10), 860–864, DOI: [10.1038/nmat3415](https://doi.org/10.1038/nmat3415).
- 8 R. Ge, X. Wu, M. Kim, J. Shi, S. Sonde, L. Tao, Y. Zhang, J. C. Lee and D. Akinwande, Atomristor: Nonvolatile Resistance Switching in Atomic



- Sheets of Transition Metal Dichalcogenides, *Nano Lett.*, 2018, **18**(1), 434–441, DOI: [10.1021/acs.nanolett.7b04342](https://doi.org/10.1021/acs.nanolett.7b04342).
- 9 M. T. Dronadula and N. R. Aluru, Solvent Structure and Dynamics Controlled Memristive Ion Transport in Å-Scale Channels, *J. Phys. Chem. Lett.*, 2026, **17**(1), 127–133, DOI: [10.1021/acs.jpcclett.5c03397](https://doi.org/10.1021/acs.jpcclett.5c03397).
- 10 M. R. Powell, L. Cleary, M. Davenport, K. J. Shea and Z. S. Siwy, Electric-Field-Induced Wetting and Dewetting in Single Hydrophobic Nanopores, *Nat. Nanotechnol.*, 2011, **6**(12), 798–802, DOI: [10.1038/nnano.2011.189](https://doi.org/10.1038/nnano.2011.189).
- 11 D. Wang, M. Kvetny, J. Liu, W. Brown, Y. Li and G. Wang, Transmembrane Potential across Single Conical Nanopores and Resulting Memristive and Memcapacitive Ion Transport, *J. Am. Chem. Soc.*, 2012, **134**(8), 3651–3654, DOI: [10.1021/ja211142e](https://doi.org/10.1021/ja211142e).
- 12 Y. Bu, Z. Ahmed and L. Yobas, A Nanofluidic Memristor Based on Ion Concentration Polarization, *Analyst*, 2019, **144**(24), 7168–7172, DOI: [10.1039/C9AN01561B](https://doi.org/10.1039/C9AN01561B).
- 13 T. Xiong, C. Li, X. He, B. Xie, J. Zong, Y. Jiang, W. Ma, F. Wu, J. Fei, P. Yu and L. Mao, Neuromorphic Functions with a Polyelectrolyte-Confined Fluidic Memristor, *Science*, 2023, **379**(6628), 156–161, DOI: [10.1126/science.adc9150](https://doi.org/10.1126/science.adc9150).
- 14 R. Song, P. Wang, H. Zeng, S. Zhang, N. Wu, Y. Liu, P. Zhang, G. Xue, J. Tong, B. Li, H. Ye, K. Liu, W. Wang and L. Wang, Nanofluidic Memristive Transition and Synaptic Emulation in Atomically Thin Pores, *Nano Lett.*, 2025, **25**(14), 5646–5655, DOI: [10.1021/acs.nanolett.4c06297](https://doi.org/10.1021/acs.nanolett.4c06297).
- 15 P. Robin, N. Kavokine and L. Bocquet, Modeling of Emergent Memory and Voltage Spiking in Ionic Transport through Angstrom-Scale Slits, *Science*, 2021, **373**(6555), 687–691, DOI: [10.1126/science.abf7923](https://doi.org/10.1126/science.abf7923).
- 16 P. Robin, T. Emmerich, A. Ismail, A. Niguès, Y. You, G.-H. Nam, A. Keerthi, A. Siria, A. K. Geim, B. Radha and L. Bocquet, Long-Term Memory and Synapse-like Dynamics in Two-Dimensional Nanofluidic Channels, *Science*, 2023, **379**(6628), 161–167, DOI: [10.1126/science.adc9931](https://doi.org/10.1126/science.adc9931).
- 17 Z. Li, S. K. Myers, J. Xiao, Y. Li, N. Noy, A. Leuski and A. Noy, Neuromorphic Ionic Computing in Droplet Interface Synapses, *Sci. Adv.*, 2025, **11**(30), eadv6603, DOI: [10.1126/sciadv.adv6603](https://doi.org/10.1126/sciadv.adv6603).
- 18 A. Ismail, G.-H. Nam, A. Lokhandwala, S. V. Pandey, K. V. Saurav, Y. You, H. Jyothilal, S. Goutham, R. Sajja, A. Keerthi and B. Radha, Programmable Memristors with Two-Dimensional Nanofluidic Channels, *Nat. Commun.*, 2025, **16**(1), 7008, DOI: [10.1038/s41467-025-61649-6](https://doi.org/10.1038/s41467-025-61649-6).
- 19 R. FitzHugh, Impulses and Physiological States in Theoretical Models of Nerve Membrane, *Biophys. J.*, 1961, **1**(6), 445–466, DOI: [10.1016/S0006-3495\(61\)86902-6](https://doi.org/10.1016/S0006-3495(61)86902-6).
- 20 J. Nagumo, S. Arimoto and S. Yoshizawa, An Active Pulse Transmission Line Simulating Nerve Axon, *Proc. IRE*, 1962, **50**(10), 2061–2070, DOI: [10.1109/JRPROC.1962.288235](https://doi.org/10.1109/JRPROC.1962.288235).
- 21 R. Ahsan, Z. Wu, S. A. A. Jalal and R. Kapadia, Ultralow Power Electronic Analog of a Biological Fitzhugh–Nagumo Neuron, *ACS Omega*, 2024, **9**(16), 18062–18071, DOI: [10.1021/acsomega.3c09936](https://doi.org/10.1021/acsomega.3c09936).
- 22 M. D. Pickett, G. Medeiros-Ribeiro and R. S. Williams, A Scalable Neuristor Built with Mott Memristors, *Nat. Mater.*, 2013, **12**(2), 114–117, DOI: [10.1038/nmat3510](https://doi.org/10.1038/nmat3510).



- 23 Y. Pei, B. Yang, X. Zhang, H. He, Y. Sun, J. Zhao, P. Chen, Z. Wang, N. Sun, S. Liang, G. Gu, Q. Liu, S. Li and X. Yan, Ultra Robust Negative Differential Resistance Memristor for Hardware Neuron Circuit Implementation, *Nat. Commun.*, 2025, **16**(1), 48, DOI: [10.1038/s41467-024-55293-9](https://doi.org/10.1038/s41467-024-55293-9).
- 24 B. Jacob, J. Silva, J. M. L. Figueiredo, J. B. Nieder and B. Romeira, Light-Induced Negative Differential Resistance and Neural Oscillations in Neuromorphic Photonic Semiconductor Micropillar Sensory Neurons, *Sci. Rep.*, 2025, **15**(1), 6805, DOI: [10.1038/s41598-025-90265-z](https://doi.org/10.1038/s41598-025-90265-z).
- 25 Y. Ma, Y. Niu, R. Pei, W. Wang, B. Wei and Y. Xie, Reconfigurable Neuromorphic Computing by a Microdroplet, *Cell Rep. Phys. Sci.*, 2024, **5**(9), 102202, DOI: [10.1016/j.xcrp.2024.102202](https://doi.org/10.1016/j.xcrp.2024.102202).
- 26 R. Yang, Y. Balogun, S. Ake, D. Baram, W. Brown and G. Wang, Negative Differential Resistance in Conical Nanopore Iontronic Memristors, *J. Am. Chem. Soc.*, 2024, **146**(19), 13183–13190, DOI: [10.1021/jacs.4c00922](https://doi.org/10.1021/jacs.4c00922).
- 27 H. J. C. Berendsen, D. van der Spoel and R. van Drunen, GROMACS: A Message-Passing Parallel Molecular Dynamics Implementation, *Comput. Phys. Commun.*, 1995, **91**(1), 43–56, DOI: [10.1016/0010-4655\(95\)00042-E](https://doi.org/10.1016/0010-4655(95)00042-E).
- 28 B. Hess, C. Kutzner, D. van der Spoel and E. Lindahl, GROMACS 4: Algorithms for Highly Efficient, Load-Balanced, and Scalable Molecular Simulation, *J. Chem. Theory Comput.*, 2008, **4**(3), 435–447, DOI: [10.1021/ct700301q](https://doi.org/10.1021/ct700301q).
- 29 S. Pronk, S. Páll, R. Schulz, P. Larsson, P. Bjelkmar, R. Apostolov, M. R. Shirts, J. C. Smith, P. M. Kasson, D. van der Spoel, B. Hess and E. Lindahl, GROMACS 4.5: A High-Throughput and Highly Parallel Open Source Molecular Simulation Toolkit, *Bioinformatics*, 2013, **29**(7), 845–854, DOI: [10.1093/bioinformatics/btt055](https://doi.org/10.1093/bioinformatics/btt055).
- 30 M. J. Abraham, T. Murtola, R. Schulz, S. Páll, J. C. Smith, B. Hess and E. Lindahl, GROMACS: High Performance Molecular Simulations through Multi-Level Parallelism from Laptops to Supercomputers, *SoftwareX*, 2015, **1–2**, 19–25, DOI: [10.1016/j.softx.2015.06.001](https://doi.org/10.1016/j.softx.2015.06.001).
- 31 S. Nosé, A Unified Formulation of the Constant Temperature Molecular Dynamics Methods, *J. Chem. Phys.*, 1984, **81**(1), 511–519, DOI: [10.1063/1.447334](https://doi.org/10.1063/1.447334).
- 32 W. G. Hoover, Canonical Dynamics: Equilibrium Phase-Space Distributions, *Phys. Rev. A: At., Mol., Opt. Phys.*, 1985, **31**(3), 1695–1697, DOI: [10.1103/PhysRevA.31.1695](https://doi.org/10.1103/PhysRevA.31.1695).
- 33 T. Darden, D. York and L. Pedersen, Particle Mesh Ewald: An $N \cdot \log(N)$ Method for Ewald Sums in Large Systems, *J. Chem. Phys.*, 1993, **98**, 10089.
- 34 H. J. C. Berendsen, J.-R. Grigera and T. P. Straatsma, The Missing Term in Effective Pair Potentials, *J. Phys. Chem.*, 1987, **91**(24), 6269–6271.
- 35 I. S. Joung and T. E. Cheatham III, Molecular Dynamics Simulations of the Dynamic and Energetic Properties of Alkali and Halide Ions Using Water-Model-Specific Ion Parameters, *J. Phys. Chem. B*, 2009, **113**(40), 13279–13290, DOI: [10.1021/jp902584c](https://doi.org/10.1021/jp902584c).
- 36 T. Liang, S. R. Phillpot and S. B. Sinnott, Parametrization of a Reactive Many-Body Potential for Mo–S Systems, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2009, **79**(24), 245110, DOI: [10.1103/PhysRevB.79.245110](https://doi.org/10.1103/PhysRevB.79.245110).
- 37 R. B. Best, X. Zhu, J. Shim, P. E. M. Lopes, J. Mittal, M. Feig and A. D. MacKerell Jr, Optimization of the Additive CHARMM All-Atom Protein Force Field Targeting Improved Sampling of the Backbone ϕ , ψ and Side-Chain X1 and



- X2 Dihedral Angles, *J. Chem. Theory Comput.*, 2012, **8**(9), 3257–3273, DOI: [10.1021/ct300400x](https://doi.org/10.1021/ct300400x).
- 38 G. Henkelman, A. Arnaldsson and H. Jónsson, A Fast and Robust Algorithm for Bader Decomposition of Charge Density, *Comput. Mater. Sci.*, 2006, **36**(3), 354–360, DOI: [10.1016/j.commatsci.2005.04.010](https://doi.org/10.1016/j.commatsci.2005.04.010).
- 39 M. Wang, X. Wang, M. Zheng and X. Zhou, Improving Catalytic Activity of “Janus” MoSSe Based on Surface Interface Regulation, *Molecules*, 2022, **27**(18), 6038.
- 40 F. Güller, A. M. Llois, J. Goniakowski and C. Noguera, Prediction of Structural and Metal-to-Semiconductor Phase Transitions in Nanoscale MoS₂, WS₂, and Other Transition Metal Dichalcogenide Zigzag Ribbons, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2015, **91**(7), 75407, DOI: [10.1103/PhysRevB.91.075407](https://doi.org/10.1103/PhysRevB.91.075407).
- 41 J. Dzubiella and J.-P. Hansen, Electric-Field-Controlled Water and Ion Permeation of a Hydrophobic Nanopore, *J. Chem. Phys.*, 2005, **122**(23), 234706, DOI: [10.1063/1.1927514](https://doi.org/10.1063/1.1927514).

