

## REVIEW

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# Advancements in transition metal dichalcogenides (TMDCs) for self-powered photodetectors: challenges, properties, and functionalization strategies

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This paper presents a comprehensive overview of the current landscape of self-powered photodetectors, emphasizing the emerging role of transition metal dichalcogenides (TMDCs) as promising materials for addressing their growing need in this field. The importance of self-powered photodetectors in a range of applications is highlighted in the introduction, which also sets the scene for the debate. The difficulties with TMDC-based photodetectors are then discussed, highlighting the necessity for creativity. A thorough investigation is conducted into the characteristics of TMDCs that are relevant to self-powered photodetectors, such as their high carrier mobilities, mechanical flexibility, thermal conductivity, strong spin-valley coupling, optical qualities, and electrical characteristics. Because of these intrinsic qualities, TMDCs are attractive options for self-powered photodetectors. This review paper then explores different functionalization approaches to induce self-powered properties in TMDCs, including surface functionalization, covalent functionalization, non-covalent functionalization, strain engineering, doping, and photoinduced charge transfer for optoelectronic property modulation. A detailed discussion of each method is provided, highlighting its potential to improve TMDCs performance. Simultaneously, various device architectures used as self-powered photodetectors, emphasizing the benefits and drawbacks of each, including avalanche photodiodes, bipolar junction transistors, and metal–semiconductor–metal (MSM) combinations have been discussed.

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# 1. Introduction

A great deal of research has been conducted on single-layer two-dimensional materials since the discovery of single-layer graphene a few years ago.<sup>1,2</sup> As a result of this interest, both known and new 2D materials were discovered, including metal dichalcogenides and boron nitrides. The creation of optical devices based on 2D materials has advanced significantly. Optoelectronic devices have shown impressive performance with the introduction of TMDCs, including strong photoluminescence, a high current ON–OFF ratio, and acceptable carrier mobility as detailed in Table 1.<sup>3</sup> Since unsaturated d-orbital transition metals are present in TMDCs, they can vary from semiconductors to semimetals:<sup>4</sup> MoS<sub>2</sub>, WS<sub>2</sub>, selenium (Se) based TMDCs like MoSe<sub>2</sub> and WSe<sub>2</sub>, and tellurium (Te) based TMDCs like

MoTe<sub>2</sub> and WTe<sub>2</sub>.<sup>5,6</sup> The type of metal and chalcogen atoms employed can affect the characteristics of TMDCs. TMDCs, like MoS<sub>2</sub> and WS<sub>2</sub>, have a layered structure with a metal atom sandwiched between two chalcogen atoms and only one or a few atomic layers of thickness.<sup>7</sup> Fig. 1 shows the periodic table highlighting the possible metals and chalcogens to form 2D TMDCs.<sup>8</sup> The study of photodetection is one of the most promising fields for TMDC research. Photodetectors are tools that turn light into electrical signals. They have a wide range of uses in fields like sensing, imaging, and telecommunications.<sup>9</sup> Because of their many benefits, self-powered photodetectors are crucial in the field of optoelectronics. By capturing and transforming incident light into electrical power, they provide energy-efficient solutions by lowering the need for external power sources like batteries and boosting energy efficiency.<sup>10</sup> By reducing electrical waste and the carbon footprint, this not only increases the operating life of electronic products but also supports environmental sustainability goals. Greater convenience in mobility can be achieved by the development of wearable, lightweight, and portable devices, which are made possible by these detectors.<sup>11</sup> Additionally, they are useful in difficult and distant locations where traditional power sources might not be available. TMDCs possess the capability to generate multiple excitons upon absorbing a single photon of light with energy exceeding the forbidden gap of materials. Photodetectors utilizing heterojunctions such as SnSe<sub>2</sub>/MoS<sub>2</sub>, MoSe<sub>2</sub>/WSe<sub>2</sub>, and WSe<sub>2</sub>/MoS<sub>2</sub> have demonstrated inherent self-driven photo-switching capabilities.<sup>12</sup> Jing Zhong *et al.* demonstrated that a photodetector utilizing a 9.5 nm thick  $\alpha$ -MoSe<sub>2</sub> thin film achieves the highest photocurrent of 4.60  $\mu$ A, a photo-to-dark current ratio (PDCR) of 3067, and a photoresponsivity of 0.94 mA W<sup>-1</sup> at zero bias.<sup>13</sup> MoS<sub>2</sub>-based photodetectors exhibit elevated photoresponsivity and rapid photo response speed, and their photon absorption is predominantly confined to the UV-visible region. This constraint arises from MoS<sub>2</sub>'s wide bandgap, approximately 1.83 eV for monolayer MoS<sub>2</sub>, limiting its utility in broadband photodetection.<sup>14</sup> Near-infrared the (NIR) photodetectors have significant importance in various military and civilian domains, encompassing optical telecommunication, target imaging, remote sensing, time surveillance, and environmental monitoring. Given its relatively small bandgap, PtSe<sub>2</sub> shows promise for potential applications in NIR detection.<sup>15</sup> 2D TMDCs emerge as promising candidates for ultralow-intensity photodetection. Nevertheless, the performance of these photodetectors is often hindered by rapid degradation induced by ambient conditions and slow response due to long-lived charge trapping, resulting in a substantial persistent photocurrent upon switching off the light source. To address these challenges, the functionalization of TMDCs with other materials is explored. Krishna Murari *et al.* developed a device using this approach by presenting an ITO/WSe<sub>2</sub>/SnSe<sub>2</sub>-based device that exhibits responsivity exceeding 1100 A W<sup>-1</sup> and achieves a rapid transient response time on the order of 10  $\mu$ s.<sup>16</sup> The strong spin valley coupling feature of TMDCs represents a strong coupling between the spin and valley states of charge carriers, presenting significant benefits.



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Table 1 Advantages of TMDCs for self-powered photodetectors

Advantages of 2D TMDCs for self-powered photo-detectors	Explanation
High electron mobility	TMDCs have high carrier mobility with a crucial property for high-speed transistors. MoS <sub>2</sub> exhibits intrinsic mobility of around 1 cm <sup>2</sup> V <sup>-1</sup> s <sup>-1</sup> when lacking high <i>k</i> dielectric gate materials. Incorporating HfO <sub>2</sub> as the top-gate layer significantly boosts this value to a promising ~150 cm <sup>2</sup> V <sup>-1</sup> s <sup>-1</sup> at room temperature. <sup>37</sup>
Mechanical flexibility	TMDCs with high mechanical flexibility empower the photodetector to form curved surfaces, expanding its adaptability for various applications such as conformal imaging devices and wearable electronics.
Thermal conductivity	The absorption of photons by the material, coupled with phonon interactions, can increase the temperature of the photodetector, inducing a conductivity change. For TMDC-based photodetectors operating exclusively in the NIR and visible spectral regimes, this effect is likely negligible. <sup>38</sup>
Strong spin valley coupling	The advantageous presence of robust spin valley coupling of TMDCs enhances the sensitivity and effectiveness of photodetectors significantly.
Wide range of optical properties	Enhanced photodetection efficiency can be readily attained through the integration of TMDCs with other semiconductors, resulting in increased optical absorption and improved charge separation
Band gap	TMDCs' variable band gaps play a key role in the absorption of light, exerting a direct influence on their optical characteristics.
Transparency	The majority of monolayer TMDCs exhibit optical transparency, having 85% transmission, with light absorption levels in the range of 5% to 10% within the visible range.

IA																		0
H	IIA											IIIA	IVA	VA	VIA	VIIA	He	
Li	Be											B	C	N	O	F	Ne	
Na	Mg	IIIB	IVB	VB	VIB	VII	VIII	IB	IIB			Al	Si	P	S	Cl	Ar	
K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr	
Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xa	
Cs	Ba	La/Lu	Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn	
Fr	Ra	Ac/Ar	Rf	Db	Sg	Bh	Hs	Mt	Ds	Rg	Uub	Uut	Fl	Uup	Lv	Uus	Uuo	

Fig. 1 Periodic table highlighting the possible metals and chalcogens to form 2D TMDCs.

It improves light-matter interactions and allows for the exact control and modification of these carriers in response to external stimuli, leading to better light absorption and photo response. A critical property of 2D materials such as TMDCs is carrier mobility, which can affect the performance of electronic components such as transistors and photodetectors.<sup>17</sup> Comparing TMDC materials to their bulk counterparts, significant carrier mobility is common. This is because smaller dimensions of the material result in less scattering of electrons and holes by phonons, impurities, and defects. Furthermore, the TMDCs' comparatively low dielectric constant makes it possible for electrons and holes to interact with one another more strongly, which improves their mobility.<sup>18</sup> Yet, several elements, including impurities, and the presence of a substrate, can have an impact on carrier mobility in TMDCs. Impurities, flaws, or a substrate can serve as electron and hole scatterers and lower carrier mobility.<sup>19</sup> Moreover, the type and thickness of a particular TMDC can affect the carrier movement. For instance, although other TMDCs like MoSe<sub>2</sub> and WSe<sub>2</sub> have lower electron mobility, MoS<sub>2</sub> and WS<sub>2</sub> have comparatively high electron mobility.<sup>20</sup> High-speed electrical devices and

high-performance optoelectronic devices like photodetectors can both benefit from the high carrier mobility of TMDCs.<sup>21</sup> To obtain high carrier mobility and maximize their performance in electrical and optoelectronic devices, quality and purity of TMDC materials must therefore be under good control.<sup>17,21</sup> TMDC materials are highly reactive and ideal for use in catalytic and sensing applications because of their huge surface area to volume ratio.<sup>22</sup> Electrons may pass through 2D materials with ease because of their high electron mobility. Due to this characteristic, TMDC materials can be used in electrical components like transistors and field-effect transistors.<sup>23</sup> Due to the reduced dimensionality of the materials and their comparatively low dielectric constant, which permits a higher Coulomb interaction between electrons and holes, this feature results.<sup>24</sup> The exciton binding energy of monolayer WS<sub>2</sub> is large (0.7 eV).<sup>25</sup> For MoS<sub>2</sub>, MoSe<sub>2</sub>, WS<sub>2</sub> and WSe<sub>2</sub> semiconductors, the theoretically expected carrier mobilities are significant (354, 269, 1739, and 1083 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> for each semiconductor).<sup>24</sup> 2D TMDCs have comparatively low hole mobility compared to their great electron mobility. Strong Coulomb interactions between holes, which result in a higher scattering





**Table 2** Carrier mobility of transition metal dichalcogenides (TMDCs) compared with wide bandgap semiconductors such as ZnO, GaN, Ga<sub>2</sub>O<sub>3</sub>, AlN, ZnS, GaAs, etc.

Materials	Charge carrier mobility (cm <sup>2</sup> V <sup>-1</sup> s <sup>-1</sup> )	Ref.
ZnO	137	58
GaN	800	59
βGa <sub>2</sub> O <sub>3</sub>	150	60
AlN	300	61
ZnS	100–300	62
GaAs	1150	63
MoS <sub>2</sub>	50–200	64
ZrTe <sub>2</sub>	1.8 × 10 <sup>4</sup> at 2 K	65
ZrSe <sub>2</sub>	2316 at room temperature	66
ZrS <sub>2</sub>	1250	67
HfS <sub>2</sub>	1800	68
HfSe <sub>2</sub>	> 2000	69
NbS <sub>2</sub>	2.68 × 10 <sup>3</sup>	70
TaTe <sub>2</sub>	1400	71
WSe <sub>2</sub>	1000	72
MoTe <sub>2</sub>	200	73
WS <sub>2</sub>	100	74

rate and a lower mobility, are to blame for this.<sup>26</sup> Due to their low dimensionality and high excitonic binding energy, 2D materials have a strong light-matter interaction. Due to this characteristic, 2D materials can be used in optoelectronic components like solar cells and photodetectors<sup>27,28</sup> as shown in Table 2. TMDCs have strong light absorption properties and can absorb up to 5–10% of incident sunlight in as little as 1 nm of thickness, several orders of magnitude higher than that of Si and GaAs.<sup>29</sup> An excellent example of this is a monolayer MoS<sub>2</sub> photodetector which can have a photoresponsivity of 880 A W<sup>-1</sup>.<sup>30</sup> Due to the anisotropic nature of 2D TMDCs' transport capabilities, multiple transport behaviors may be seen depending on the direction in which a material is measured. 2D TMDCs' electronic characteristics can be adjusted using this anisotropy to tailor them to certain uses. Transport features of 2D TMDCs are temperature-dependent, which means that as the temperature changes, so can the behavior of the devices. If the temperature drops, for instance, the electron mobility of 2D TMDCs typically rises while the hole mobility falls.<sup>31</sup> A substantial anisotropy in the electrical and optical characteristics of TMDC materials can result from broken inversion symmetry, and this anisotropy affects the behavior of the material.<sup>32</sup> TMDC materials have a large excitonic binding energy, causing the creation of excitons to result in a significant absorption of light (bound electron-hole pairs). It is possible to exploit this attribute to modify the optical characteristics of TMDC materials. 2D TMDCs exhibit a quantum confinement effect, which means that the materials' reduced dimensionality affects transport characteristics of the material. The electronic band structure and electronic density of states may change as a result of these quantum confinement effects, which may change the transport properties of the material.<sup>33</sup> Other types of scattering, such as scattering by phonons, impurities, and defects, can also have an impact on the transport characteristics of 2D TMDCs. These scattering processes have the potential to change the electrical band structure as well as the mobility of carriers. Because of their

many unique qualities, these materials may be used in a wide range of applications in electronics, optoelectronics, and spintronics. These qualities may be controlled and altered by varying the synthesis conditions, material thickness, and the kind of metal and chalcogen atoms used. These properties are essential for comprehending the electronic behavior of 2D TMDCs and enhancing their performance in electrical devices.

Transition metals and chalcogen atoms are shown in Fig. 1 by highlighted colors. Materials based on the four group IV(d2), V(d3), VI(d4), and VII(d5) metals form a layered structure, while non-layered materials are usually derived from the eighth group VIIIA–VIIIc (d6–d8) metals. It is easy to modulate the photoelectric response of TMDCs, regardless of the semiconductor or near metal they are made from. The number of layers in TMDC materials influences their optical properties based on their energy band structures. Since nanopores based on the atomically thin medicine of TMDCs may give a quick, accurate approach for high-resolution DNA sequencing, it has been demonstrated that TMDCs offer great potential for biophysical applications, including DNA sequencing and personalized medicine.<sup>19</sup> A photodetector is a device capable of transforming an electrical signal into an optical signal. In recent times, numerous researchers have made significant advancements in the field by creating photodetectors that rely on transition metal dichalcogenides. In 2017, Wenzhi Yu *et al.* introduced near-infrared photodetectors that utilize a MoTe<sub>2</sub>/graphene heterostructure, and these photodetectors exhibit remarkable performance matrices, including a notably high responsivity of approximately 970.83 A W<sup>-1</sup>, a substantial photoconductive gain of 4.69 × 10<sup>8</sup>, and an impressive detective value of 1.55 × 10<sup>11</sup> cm Hz<sup>1/2</sup> W<sup>-1</sup>.<sup>34</sup> In 2019, Pratik Pataniya *et al.* introduced a paper-based flexible photodetector that has enhanced the functionalization of WSe<sub>2</sub> nanodots. This photodetector demonstrates an outstanding response time of 0.68 s. Additionally, it boasts a high responsivity value of 17.78 mA W<sup>-1</sup>, along with a detectivity of 5.68 × 10<sup>10</sup> Jones.<sup>35</sup> The majority of photodetectors necessitate an external power source to facilitate the movement of the charge carrier. Unfortunately, this requirement contributes to the enlargement of the overall device size and enforces a dependence on additional external equipment. Self-powered photodetectors represent a category of devices that operate without the need for an external power supply. These photodetectors have the potential to hold significant importance in applications such as the Internet of Things and optical sensor networks.<sup>36</sup>

This review aims to comprehensively estimate the performance, characteristics, and applications of self-powered photodetectors based on TMDCs. Additionally, it identifies current challenges and outlines future prospects for this promising technology. The properties of TMDCs that make them suitable for self-powered photodetectors have been explained, along with an exploration of how these properties can be enhanced through various functionalization strategies. It explores into the optical, electrical, and electronic properties of TMDCs and explores their societal impact. To gain a better understanding and enhance the overall performance of the photodetector, we





provide an in depth explanation of the device structure of the self-powered photodetector and how these materials are used in different types of photodetectors like phototransistors and photodiodes. Additionally, we have presented various TMDC-based self-powered photodetectors, organized in table form, highlighting their parameters.

## 2. Need for self-powered photodetectors

Photodetectors are devices which transform light into an electrical signal. Self-powered photodetectors represent a novel category of photodetectors that facilitate photodetection without relying on external power sources.<sup>39</sup> This concept has gained momentum due to the increasing popularity of wearable photodetectors. They are utilized in numerous fields, such as data storage, sensing, imaging, and communications. The capacity of photodetectors to detect light with high sensitivity and low noise is one of their key features.<sup>7,27,31</sup> To detect and enhance light signals conveyed across optical fibers, photodetectors are employed in fiber-optic communication systems. Digital cameras, camcorders, and other imaging equipment employ photodetectors to turn light into an electrical signal that can be processed and saved as an image.<sup>40</sup> UV communication has significant advantages over optical wireless communication methods due to its inherent benefits of resistance to interference and interception. It appears that using self-powered photodetectors as optical receivers will improve the integration and mobility of UV communication systems.<sup>41</sup> A variety of sensing applications, including gas sensing, environmental sensing, and biometric sensing, use photodetectors.<sup>42,43</sup> Flexible self-powered photodetectors are a significant component of flexible self-powered electronic systems, encompassing biosensors, nanogenerators, and photodetectors. Notably, these systems operate without the requirement for external energy sources, utilizing the photovoltaic effect even at zero voltage. This eliminates the necessity of developing a supplementary power supply system.<sup>44</sup>

Photodetectors are used to read and write data on data storage devices such as CDs, DVDs, and Blu-ray discs. Using graphene, Mohit *et al.* developed photon-triggered, self-biased electronic circuits in 2021.<sup>45</sup> These circuits included a variety of parts, including switches, transistors, logic gates, and lateral p-n and n-p homojunctions. Local probe measurements validated these advances.<sup>45</sup> To transform X-ray and gamma-ray photons into an electrical signal, photodetectors are employed in medical imaging devices like X-ray and CT scanners. Photodetectors are utilized in systems like motion sensors and burglar alarms. Photodetectors are used in spectroscopy to find the light that a sample emits or absorbs. Self-powered ultraviolet photodetectors hold immense promise in optical communication, smart security systems, and space exploration by enabling photodetection without the need for external power sources. In conclusion, photodetectors are crucial components of contemporary technology. From Fig. 2, we can see that they



Fig. 2 Applications of TMDC photodetector devices.

are widely utilized in a variety of applications that call for the detection and conversion of light into an electrical signal. 2D TMDCs have been investigated as photodetectors due to their unique optical and electrical properties, such as their high absorption coefficient, high on/off ratio, and high responsiveness.<sup>46</sup>

## 3. Current challenges and potential solution to address these challenges in TMDCs for photodetectors

Photodetectors must be able to recognize weak signals while using the least amount of power possible.<sup>47</sup> Fig. 3 represents the properties and challenges associated with TMDCs. Photodetectors must be able to detect a variety of light intensities without being saturated. Also, they recognize and react to quickly changing lighting conditions. Photodetectors must be able to function under a variety of circumstances without deteriorating over time. Researchers are creating new kinds of photodetectors that may perform better than conventional silicon-based devices. Examples of these include those based on novel materials. For mass manufacture and commercialization, photodetectors must be reasonably priced and cost-effective.<sup>48</sup> TMDCs can be readily synthesized by mechanically transferring and stacking various combinations of TMDC material heterojunctions. The absence of lattice mismatch issues, attributed to the van der Waals forces between layers, coupled with a more complex energy band structure, enhances the potential of photodetectors based on TMDC heterojunctions, offering extensive application possibilities. To enhance the photocurrent, the conductivity of TMDCs can be elevated through methods such as doping or creating heterojunctions with compatible conductive materials. Enhancing the photoluminescence intensity and quality of monolayer TMDCs presents a noteworthy challenge, critical for achieving low threshold lasers and high quantum efficiency LEDs. The examination of excitonic dynamics, which shows the fundamental mechanisms leading to excitonic formation and emission





Fig. 3 Properties and exciting challenges associated with TMDCs.

processes, is considered an effective tool for evolving the optical properties of TMDCs.<sup>49</sup> Conventional photodetectors employing photosensitive materials face several challenges, including lattice mismatch, complex engineering processes, difficulties in incorporation with complementary metal oxide semiconductor devices, issues of toxicity, and high costs.<sup>50</sup> The discovery and utilization of TMDCs in optoelectronic devices have prompted the development of a growing array of photodetectors using these materials. These advancements, enabled by diverse technology, have showcased outstanding performance skills.

## 4. Properties of TMDCs relevant to self-powered photodetectors

### 4.1. High carrier mobilities

TMDCs with high carrier mobilities are essential for self-powered photodetectors because they facilitate effective charge transport, minimize carrier recombination, improve responsivity, speed up response times, and reduce noise, all of which improve signal-to-noise ratios.<sup>51,52</sup> Together, these characteristics make photodetectors extremely responsive, sensitive, and able to precisely transform incident photons into electrical signals. They are therefore ideal for many uses, including optical communication, sensing, and imaging systems. Carrier mobility, which gauges how quickly electrons or holes may pass through a substance, is a critical component in determining how effectively electrical devices work.<sup>53</sup> In 2D TMDCs, the carrier mobility is influenced by impurities as well as the strength of the contacts between the carriers and the lattice of the material.<sup>54</sup> Due to the presence of a suitable bandgap in the material, 2D TMDCs typically have substantially higher carrier mobility than other two-dimensional materials like graphene. High on/off ratios are produced in field-effect transistors as a result, enabling effective field-effect control of the carrier density. Reduced impurities and defects in the material as well as the use of substrates with high levels of crystalline perfection can further increase the carrier mobility of 2D TMDCs. For instance, minimizing the contact between the 2D material and the substrate by employing hexagonal boron nitride (h-BN) as a substrate can dramatically increase the carrier

mobility of MoS<sub>2</sub>.<sup>55</sup> For 2D TMDCs, the maximum carrier mobility reported is typically in the range of  $104 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  for MoS<sub>2</sub>, WSe<sub>2</sub>, and MoSe<sub>2</sub> and  $103 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  for WS<sub>2</sub>.<sup>56</sup> These values are far greater than those of other 2D materials like graphene, but they are still less than those of traditional semiconductors like silicon and GaAs. Carrier mobilities of some TMDCs are given in Table 2. It is important to note that temperature, humidity, and strain can all have an impact on how mobile TMDCs are. The quality of the samples also plays a role; samples developed using MBE or CVD typically have higher mobilities than samples grown using exfoliation or solution-based techniques.<sup>5,24,33</sup> Fig. 4(a–c) depict the distribution of mobility derived using peak transconductance for various channel widths of MoS<sub>2</sub> and WS<sub>2</sub>, respectively. The schematic depiction of PMMA (poly methyl methacrylate) aided wet transfer of monolayer TMDC is shown together with its device construction and SEM picture.<sup>57</sup> When the channel width is scaled from  $5 \mu\text{m}$  to  $100 \text{ nm}$ , the mobility of both materials varies from  $23.9 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  to  $3.6 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  for MoS<sub>2</sub> and  $29 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  to  $2.7 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  for WS<sub>2</sub>. Both materials display considerable reliance on the channel width. For MoS<sub>2</sub> and WS<sub>2</sub>, Fig. 4(d and e) display the total resistance with respect to the channel width, and Fig. 4(f and g) display the distribution of contact resistance. Contact-gating is responsible for a reduction in contact resistance with increased carrier density ( $n_s$ ), since the back-gate voltage modulates the Schottky barrier (SB) width at the metal/2D interface. The combined contribution of channel resistance and contact resistance with respect to channel width is shown in Fig. 4(h and i).

### 4.2. Mechanical flexibility

Mechanical flexibility is crucial for self-powered photodetectors for a number of reasons. Firstly, it enables the photodetector to adapt to curved or uneven surfaces, which increases its versatility for a range of uses, including conformal imaging devices and wearable electronics.<sup>75</sup> The ability of photodetectors to withstand mechanical stress or strain during handling or operation can be enhanced by mechanical flexibility. For flexible displays and medical sensors, for instance, where the photodetector may be twisted or stretched, this characteristic is





Fig. 4 (a) Schematic representation of PMMA assisted wet transfer of monolayer TMDCs with their device structure and SEM image. (b) and (c) Distribution of mobility extracted using peak transconductance for different channel widths of MoS<sub>2</sub> and WS<sub>2</sub>. (d) and (e) Total resistance with respect to the channel width for MoS<sub>2</sub> and WS<sub>2</sub>. (f) and (g) Distribution of contact resistance for MoS<sub>2</sub> and WS<sub>2</sub>. (h) and (i) Total contribution of contact resistance and channel resistance with respect to the width of the channel. Reprinted with permission from ref. 57 Copyright of Nature 2021.

quite helpful.<sup>76,77</sup> All things considered, mechanical flexibility is an essential characteristic that raises the robustness and potential uses of photodetectors that run on their own energy. 2D TMDCs are known for their mechanical flexibility,<sup>78</sup> or their capacity to bend and deform without breaking.<sup>79</sup> Weak van der Waals interactions between layers of the material are to blame for this characteristic. By evaluating the flexibility of products created from the material, such as transistors or solar cells, it is possible to determine the mechanical flexibility of 2D TMDCs. The bending radius or maximum strain that these devices can sustain before breaking or performing poorly is often used to gauge their flexibility. Thinner layers are more flexible than larger ones, hence the mechanical flexibility of 2D TMDCs is also correlated with their thickness.<sup>80</sup> For instance, although multi-layer MoS<sub>2</sub> can withstand up to 0.5% strain, a single-layer MoS<sub>2</sub> has been claimed to be flexible up to 1.5% strain.<sup>81</sup> The WSe<sub>2</sub> photodetector with graphite electrodes and h-BN encapsulation is schematically shown in Fig. 5(a). The optical image of the device is shown in Fig. 5(b), and the HR-TEM cross-sectional image is shown in Fig. 5(c). The illumination of the light on the WSe<sub>2</sub> photodetector is shown in Fig. 5(d). Yixuan Zou *et al.* investigated the photoelectric properties of WSe<sub>2</sub> photodetectors.<sup>82</sup> The  $I_{ds}$ - $V_{gs}$  transfer curve changed higher over a range of gate bias values when exposed to white light at 20 °C, showing positive photoconductivity (PPC), as seen in Fig. 5(e). Remarkably, white light at 400 °C shifted the transfer curve to the left, with the shift increases with increasing light

intensity; this is seen in Fig. 5(f). PPC was more prevalent in the N branch compared to NPC (negative photoconductivity) in the P branch. At 400 °C,  $I_{ds}$  rose by setting  $V_{gs}$  at 3 V, converting the WSe<sub>2</sub> device into a PPC photodetector (Fig. 5d). On the other hand,  $I_{ds}$  dropped at 0 V, indicating that it was an NPC photodetector (Fig. 5e). The substrate on which the material is formed and the technique utilized to manufacture the material can both have an impact on the mechanical flexibility of 2D TMDCs. For instance, flexibility can be increased by growing it on a flexible substrate like polyethylene terephthalate (PET).<sup>81</sup> A wide variety of flexible electronic applications, including flexible sensors, displays, and energy storage devices, can benefit from the mechanical flexibility of 2D TMDCs.<sup>83</sup> Furthermore, stretchable and wearable electronic devices can be created using flexibility of TMDCs. It is important to note that building heterostructures of TMDCs, where the mechanical characteristics of the layers may be modified by altering the composition of the layers, can also improve the mechanical flexibility of TMDCs.

#### 4.3. Thermal conductivity

The characteristic that characterizes the capacity to conduct heat is called thermal conductivity. It affects how heat is dissipated during operation, which affects the device's thermal management. Maintaining the photodetector's stability and performance requires effective heat control. Elevated thermal conductivity can disperse surplus heat, averting overheating





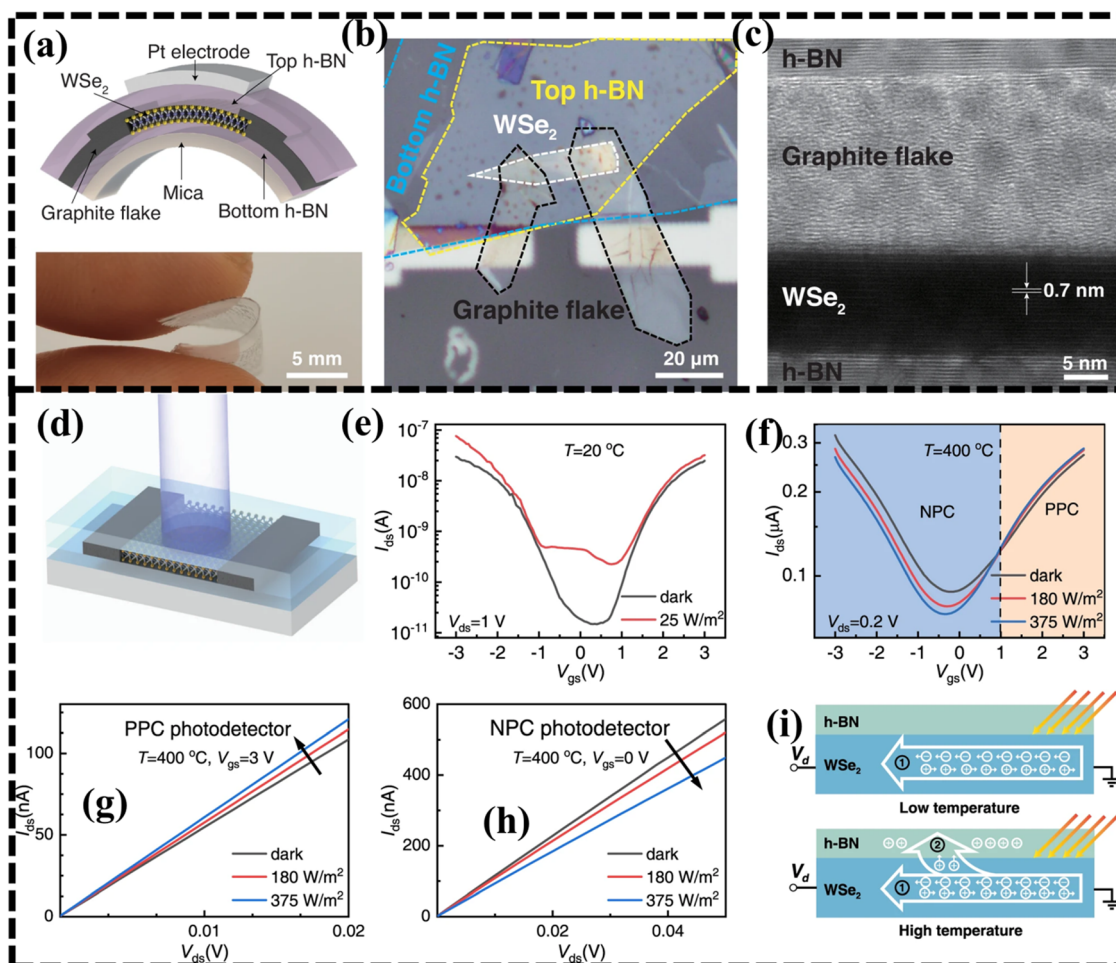


Fig. 5 (a) Schematic view of the WSe<sub>2</sub> photodetector with h-BN encapsulation and graphite electrodes. (b) Optical image of the device. (c) HR-TEM cross-sectional image of the device. (d) WSe<sub>2</sub> photodetector under illumination. (e)  $I_{ds}$ - $V_{gs}$  characteristics under dark and illumination at 20 °C. (f)  $I_{ds}$ - $V_{gs}$  characteristics under dark and illumination at 400 °C. (g)  $I_{ds}$ - $V_{ds}$  characteristics  $V_{gs} = 3$  V. (h)  $I_{ds}$ - $V_{ds}$  characteristics  $V_{gs} = 0$  V and at 400 °C. (i) Schematic view of the mechanism for the photodetection. Reprinted with permission from ref. 82 Copyright of Nature 2022.

and possible harm to the apparatus.<sup>84</sup> The photodetector's total energy efficiency may be impacted by thermal conductivity. High thermal conductive materials may effectively move heat away from the active areas of the device, minimizing energy losses from unintended heat dissipation and enhancing overall performance.<sup>85</sup> Certain devices, like thermo photodetectors, may purposefully alter their heat conductivity in order to improve their functionality. These devices work by turning incoming photons into heat, which is subsequently recognized as an electrical signal. As seen in Fig. 6(a), FETs on CVD-grown monolayer MoS<sub>2</sub> are constructed in a back-gated form. Fig. 6(b) depicts the back-gated transistor as it was manufactured. The Raman spectra show that the in-plane ( $E_{2g}$ ) and out-of-plane ( $A_{1g}$ ) Raman modes are separated by around 19 cm<sup>-1</sup>, as seen in Fig. 6(c). This is done to confirm that the channel is thick MoS<sub>2</sub> in a single layer. Optimizing the efficiency of these detectors requires controlling the thermal conductivity. The internal phonon scattering processes in 2D TMDCs control the thermal conductivity of the substance.<sup>86</sup> Due to the weak van der Waals forces holding the layers together, 2D TMDCs have a

lower thermal conductivity than bulk materials. The out-of-plane thermal conductivity, which is typically in the range of 1–5 W m<sup>-1</sup> K<sup>-1</sup>, plays a major role in explaining the poor thermal conductivity of 2D TMDCs.<sup>87</sup> Defects, contaminants, and strain can all be added to a material to further diminish its thermal conductivity. For thermoelectric applications, where a low thermal conductivity is sought to increase the thermoelectric figure of merit, this may be advantageous.<sup>88</sup> Moreover, the crystalline quality can be increased, the number of layers can be decreased, and substrates with high thermal conductivity, like h-BN, can be used. It is important to note that temperature, humidity, and strain can all have an impact on the thermal conductivity of 2D TMDCs. The quality of the samples can also have an impact on the thermal conductivity of 2D TMDCs; samples developed by CVD or MBE often have higher thermal conductivities than samples formed by exfoliation or solution-based techniques. Guangqian Ding *et al.* investigated the Seebeck coefficient ( $S$ ) for monolayer ZrSe<sub>2</sub> and HfSe<sub>2</sub> at various temperatures (400, 600, 800, and 1000 K).<sup>89</sup> At lower temperatures, both materials show identical  $S$  values;





**Fig. 6** (a) Schematic diagram of a back-gated transistor. (b) Optical image of the photodetector device. (c) Raman spectrum of the MoS<sub>2</sub> channel with a peak separation of 19 cm<sup>-1</sup> confirming the monolayer. Reprinted with permission from ref. 90. Copyright of Nature 2023. (d) The temperature dependence of in-plane thermal conductivity of LA, TA and ZA branches of monolayer WSe<sub>2</sub>. (e) The relative contribution of spectral phonons to in-plane thermal conductivity in monolayer WSe<sub>2</sub>. Reprinted with permission from ref. 91 Copyright of Nature 2015.

however, a bipolar effect is seen at higher temperatures and lower carrier concentrations. Excitable electrons cause  $S$  peaks to drop and shift in the direction of increasing carrier concentrations. For a wide range of carrier concentrations, both materials show high  $S$  values ( $>200 \mu\text{V K}^{-1}$ ), with n-type  $S$  showing some slight superiority. Monolayer HfSe<sub>2</sub> exhibits a higher  $S$  value of  $660 \mu\text{V K}^{-1}$  at 600 K. Power factors indicate that n-type doping leads to improved thermoelectric performance, with HfSe<sub>2</sub> beating ZrSe<sub>2</sub>. In the range of ideal carrier concentration, these power factors exhibit modest temperature dependency, suggesting that they are temperature-adaptable. While bulk ZrSe<sub>2</sub> and HfSe<sub>2</sub> show larger power factors with narrower band gaps and lower optimum carrier concentrations, monolayer MoSe<sub>2</sub> and ZrSe<sub>2</sub> have similar electrical transport capabilities. In Fig. 6(d), the thermal conductivities of the LA, TA, and ZA branches in WSe<sub>2</sub> with a 1  $\mu\text{m}$  size demonstrate a proportional relationship at low temperatures due to linear

dispersion close to the  $\Gamma$ -point. However, at high temperatures, they exhibit a  $1/T$  behavior that points to phonon-phonon scattering. Fig. 6(e) displays the contributions of spectral phonons to thermal conductivity in monolayer WSe<sub>2</sub>. The contributions of ZA phonons are the most prominent and increase with temperature. At room temperature, ZA phonons provide approximately 80% more than monolayer MoS<sub>2</sub>.

#### 4.4. Strong spin-valley coupling

For materials such as TMDCs, strong spin-valley coupling is an essential characteristic, especially in the field of self-powered photodetectors.<sup>92</sup> Strong spin-valley coupling is beneficial for creating extremely sensitive and effective photodetectors because of this feature as well as its potential for valleytronics applications and high quantum efficiency.<sup>93</sup> Due to the reduced dimensionality of the material and the presence of a substrate in 2D materials, broken inversion symmetry can happen.



Inversion symmetry breaking in 2D materials has significant effects on such electrical and optical characteristics.<sup>94</sup> For instance, it results in a phenomenon known as a strong spin-valley coupling, where the degrees of freedom associated with spin and valley of an electron are closely connected. Due to the high levels of spin polarization and valley polarization produced by this strong spin-valley coupling, 2D materials are well suited for spintronic applications including magnetic memory and logic circuits. For instance, the electrical anisotropy of 2D materials can be exploited to regulate the functionality of electronic components like transistors and photodetectors.<sup>95</sup> In conclusion, broken inversion symmetry is a crucial characteristic of 2D materials that may be exploited to tailor and control their electrical and optical characteristics for a variety of applications in electronics, optoelectronics, and spintronics.

#### 4.5. Optical properties

In the field of self-powered photodetectors, optical characteristics play a crucial role in determining how materials interact with light.<sup>96</sup> These characteristics include the bandgap energy, refractive index, photoluminescence, photoconductivity, absorption spectrum, nonlinear optical phenomena, and polarization sensitivity. Photodetectors' sensitivity to various light wavelengths may be precisely tuned by adjusting their bandgap energy and absorption spectrum, which makes them sensitive to certain light sources and applications.<sup>97</sup> Optimizing light absorption with techniques like total internal reflection or antireflection coatings requires control over the refractive index. Moreover, photoluminescence may be used for sensitive light detection, and photoconductivity allows absorbed photons to be converted into electrical impulses. Advanced signal processing can be facilitated by nonlinear optical phenomena, and polarization sensitivity makes it possible to identify polarized light sources. Self-powered photodetectors require careful design and performance optimization of these optical features for a range of applications. For a variety of optoelectronic applications, the remarkable optical properties of MoS<sub>2</sub>, WS<sub>2</sub>, and MoSe<sub>2</sub> have sparked a lot of interest.<sup>98</sup> Because of their unique optical characteristics, these materials are interesting choices for a variety of optoelectronic applications, including solar cells, photodetectors, LEDs, and more. Because of strong absorption coefficients of MoS<sub>2</sub> in the visible and near-infrared parts of the spectrum, it is a great option for solar cells and photodetectors. Because of its high excitonic binding energy, light is significantly absorbed, especially in the near-infrared range, and excitons are produced. Because of its intense photoluminescence (PL), MoS<sub>2</sub> is also an excellent material to employ in light-emitting diodes (LEDs) and other optoelectronic devices. Polarization dependence and anisotropy enable it to have unique optical features. Two optical properties shared by WS<sub>2</sub> and MoS<sub>2</sub> include strong absorption in the visible and near-infrared spectra and a high excitonic binding energy leading to considerable light absorption. A further characteristic of WS<sub>2</sub> that makes it a desirable choice for LEDs and other optoelectronic devices is its strong photoluminescence in the visible region. Strong anisotropy and

polarization dependency, which are characteristics of its optical properties, allow for the customization of its optical behavior.<sup>99</sup> Notably, because of its strong excitonic absorption in the near-infrared spectrum, WS<sub>2</sub> is especially well-suited for near-infrared imaging and detection. MoSe<sub>2</sub> performs exceptionally well in optoelectronic applications and provides a wide range of optical characteristics. Maciej R. Molas *et al.* reported room-temperature photoluminescence (PL) and reflectance contrast (RC) spectra of an exfoliated MoS<sub>2</sub> monolayer before and after four successive bis(trifluoro methane) sulfonamide (TFSI) passivation such as shown in Fig. 7(a–c).<sup>100</sup> The first passivation leads to a two-fold increase in PL intensity, a 30 meV blue shift of the A exciton emission line as in Fig. 7(a), and similar blue shifts in RC spectra for A- and B-exciton resonances, accompanied by increased intensity and reduced linewidth. These changes illustrate a significant impact of super acid treatment on the optical properties of MoS<sub>2</sub>. Fig. 7d displays normalized PL spectra obtained from distinctively dosed areas within a monolayer MoSe<sub>2</sub> crystal. The green trace represents an unaltered region of the crystal, exhibiting distinct peaks at approximately 1.55 eV and ~ 1.77 eV, corresponding to the A- and B-excitations, respectively.<sup>101</sup> MoSe<sub>2</sub>, like MoS<sub>2</sub> and WS<sub>2</sub>, has strong photoluminescence and a high excitonic binding energy, which make it useful in optoelectronic devices. Numerous methods, including chemisorption and physisorption doping, can be used to modify intensity of the photoluminescence and spectral properties. The material's optical band gap may change with temperature. Hsiang-Lin Liu *et al.* discovered that when temperature increased, the resonance linewidth and peak locations of A and B excitons in monolayer TMDCs expanded to lower energies.<sup>102</sup> An exciton peak location was used to calculate the optical band gap, which is the amount of energy required to produce an exciton by optical absorption. Variations in chemical composition, including electronegativity, caused this gap to fluctuate. It dropped as the temperature increased. The Bose–Einstein model explains this redshift in the optical band gap with increasing temperature. The band gap of TMDCs ranges from 1.0 to 2.1 eV; presented in Table 3.

#### 4.6. Electronic properties

Fundamental characteristics of materials called electronic properties have a significant influence on the effectiveness and operation of self-powered photodetectors.<sup>105</sup> These attributes cover a wide range of traits pertaining to the behavior of electrons, including energy levels, band structure, doping, recombination processes, carrier lifespan, electron and hole mobility, and density of states. Quantum efficiency may be maximized by comprehending recombination processes, and doping can be used to adjust conductivity and spectral sensitivity. Additionally, effective charge extraction and device design depend on an understanding of energy levels and band alignment. The development of high-performance self-powered photodetectors depends critically on tailoring these electrical features to the unique needs of photodetection applications. MoS<sub>2</sub>, WS<sub>2</sub>, and MoSe<sub>2</sub> are examples of TMDCs, a family of materials recognized for their varied and useful electrical







Fig. 7 Effect of successive bis(trifluoroethane) sulfonamide passivation on (a) photoluminescence and (b) reflectance spectra of the MoS<sub>2</sub> monolayer. Reprinted with permission from ref. 100, Copyright of Nature 2019. (c) Photoluminescence (PL) photoluminescence excitation (PLE) and absorbance characterization of WS<sub>2</sub>. Reprinted with permission from ref. 103, Copyright of Nature 2019. (d) PL spectra of MoSe<sub>2</sub> corresponding to the A exciton and B exciton. Reprinted with permission from ref. 101, Copyright of Nature 2016. (e)–(h) Temperature-dependent optical band gap of monolayer MoS<sub>2</sub>, MoSe<sub>2</sub>, WS<sub>2</sub>, and WSe<sub>2</sub>: spectroscopic ellipsometry and first-principles calculations. Reprinted with permission from ref. 102, Copyright of Nature 2020.

Table 3 The optical band gap of transition metal dichalcogenides<sup>104</sup>

	S monolayer	S bulk	Se monolayer	Se bulk	Te monolayer	Te bulk
Mo	1.8–2.1	1.0–1.3	1.4–1.7	1.1–1.4	1.1–1.3	1.0–1.2
W	1.8–2.1	1.3–1.4	1.5–1.7	1.2–1.5	~1.03	Metal
Ti	~0.65	~0.3	~0.51	Metal	~0.1	Metal
Zr	~1.2	~1.6	~0.7	~0.8	~0.4	Metal
Hf	~1.3	~1.6	~0.7	~0.6	~0.3	Metal
V	~1.1	Metal	Metal	Metal	Metal	Metal
Nb	Metal	Metal	Metal	Metal	Metal	Metal
Ta	Metal	Metal	Metal	Metal	Metal	Metal
Ni	~0.6	~0.3	~0.12	Metal	Metal	Metal
Pd	~1.2	~1.1	~1.1	~1.3	~0.3	~0.2
Pt	~1.9	~1.8	~1.5	~1.4	~0.8	~0.8

characteristics. For example, MoS<sub>2</sub> has a distinct set of properties that affect its electrical conductivity, such as a band gap that is 1.8 eV for single-layer versions and 1.2 eV for multilayer ones. In addition, its high electron mobility allows for effective electron flow, which makes it a perfect fit for electrical components like transistors.<sup>106</sup> Its strong light-matter interaction also makes it suitable for use in light-emitting diodes (LEDs) and photodetectors due to its low dimensionality and high excitonic binding energy. Because of its significant anisotropy, which allows for customization of its electrical and optical characteristics, MoS<sub>2</sub> exhibits great potential in the field of spintronics.<sup>80</sup> Similarly, band gaps and electron mobility are two electrical properties that WS<sub>2</sub>, another TMDC, has in common with MoS<sub>2</sub>.<sup>107</sup> But unlike MoS<sub>2</sub>, WS<sub>2</sub> shows a clear direct band gap of 2.0 eV for single-layer WS<sub>2</sub> and an indirect band gap of 1.3 eV.<sup>77</sup> Due to its high electron mobility, it may be used in

electrical components such as field-effect transistors and transistors. Because of its high excitonic binding energy and low dimensionality, WS<sub>2</sub> exhibits a strong light-matter interaction, making it suitable for use in LEDs and photodetectors. Similar to MoS<sub>2</sub>, WS<sub>2</sub> has a strong spin-valley coupling and is adaptable for a range of applications due to its ability to change its electrical and optical characteristics depending on anisotropy. Furthermore, MoSe<sub>2</sub>, with its electrical characteristics, broadens the range of TMDCs. Electrical properties of MoSe<sub>2</sub> are influenced by its direct and indirect band gap features. For a variety of applications, it is essential to comprehend these band gaps' locations and how they affect conductivity. Like its siblings, MoSe<sub>2</sub> demonstrates anisotropy in its optical and electrical characteristics, significant spin-valley coupling, and light-matter interactions. All things considered, these TMDCs, MoS<sub>2</sub>, WS<sub>2</sub>, and MoSe<sub>2</sub>, with their particular and customized electrical properties, lead to a multitude of opportunities in electronics, optoelectronics, and spintronics, each providing special benefits and features for certain applications. As shown in Fig. 8(a), Xin-Gang Zhao *et al.*<sup>108</sup> were systematically studied three forms of heterostructured superlattices [(A)<sub>n</sub>-(B)<sub>10-n</sub>]: type A, which consists of A and B with the same transition metal M but different chalcogens X; type B, which consists of A and B with the same chalcogen X but different transition metal M; and type C, which consists of A and B with distinct chalcogen X and transition metal M, within 10-layer periodic unit cell. Fig. 8(b) and (d) show the band structures of 10-layer stacked MoSe<sub>2</sub> and MoTe<sub>2</sub>. These band structures show indirect band-gaps. Conversely, a [(MoSe<sub>2</sub>)<sub>2</sub>-(MoTe<sub>2</sub>)<sub>8</sub>] heterostructure is



Fig. 8 (a) A schematic 10-layer unit cell to construct heterostructured superlattices, (b)–(d) band structure of  $\text{MoSe}_2$ ,  $(\text{MoSe}_2)_2-(\text{MoTe}_2)_8$  and  $\text{MoTe}_2$  supercells, (e) direct and indirect band gaps depending on the content. Reprinted with permission from ref. 108, Copyright of Wiley 2020, (f) schematic illustration of room temperature iodine-doped synthesis of CuS with their AFM surface topology, (g) CuS-I/ $\text{MoS}_2$  phototransistor device, (h) drain current of the device, (i) the photoresponsivity of the device, and (j) photoresponsivity with respect to gate voltages. Reprinted with permission from ref. 109, Copyright of Wiley 2023.

depicted in Fig. 8(c) that displays a direct bandgap at the  $K$  point, signifying a transition from an indirect to a direct bandgap. As seen in Fig. 8(e), direct bandgaps of 0.27 and 0.32 eV are obtained when  $n$  is 1 and 2 in  $[(\text{MoSe}_2)_n-(\text{MoTe}_2)_{10-n}]$ . Fig. 8(f) shows the schematic illustration of room temperature iodine-doped synthesis of CuS with their AFM surface topology. Taehun Kim *et al.* fabricated a CuS-I/ $\text{MoS}_2$  phototransistor with reduced contact barriers demonstrating enhanced phototransistor performance such as shown in Fig. 8(g).<sup>109</sup> Its photoresponsivity exhibits a linear relationship with incident power and surpasses the CuS/ $\text{MoS}_2$  phototransistor under all gate biases (Fig. 8h–j). Notably, a high photoresponsivity of  $2.25 \times 10^4 \text{ A W}^{-1}$  is achieved, and a maximum detectivity ( $D^*$ ) of  $1.16 \times 10^{11} \text{ Jones}$  is attained under negative back-gate voltages.

## 5. Functionalization strategies to induce self-powered properties

By combining them, atom-thin TMDC materials may be produced separately and layered to create van der Waals-bonded heterostructures, which open up surprisingly new possibilities for functional devices.<sup>110</sup> Functionalization means the modification of the surface and structure of these materials with various functional groups or molecules. Parameters of a photodetector such as responsivity and detectivity are enhanced in these materials after functionalizing them with other suitable

materials. For TMDCs to be incorporated into future electronic devices, doping becomes a crucial first step.<sup>111</sup>

### 5.1. By doping

TMDC-based photodetectors using doping can be used to provide self-powered properties through optoelectronic behavior and electronic structure modification of the material.<sup>1</sup> TMDCs may be doped to become either n-type (electron-donor) or p-type (hole-donor) by carefully adding certain impurities or foreign atoms. This will change their carrier concentration and produce an inherent potential. Because of its intrinsic potential, the photodetector can produce electricity on its own when light strikes it, making it a self-sufficient gadget that does not require an external power source.<sup>112</sup> The required doping effects may be achieved by a variety of doping processes, including chemical vapor deposition, ion implantation, and chemical functionalization. This versatility makes TMDCs an excellent platform for the development of self-powered photodetectors with customized optoelectronic features. Noble metals are particularly advantageous due to their resistance to environmental corrosion and oxidation, making them ideal candidates for enhancing the properties of TMDCs.<sup>113,114</sup> Stable doping is required for the incorporation of TMDCs into durable device manufacturing. Hu *et al.* verified that an electric field may adjust the Schottky barrier at the graphene/ $\text{MoS}_2$  interface, converting the n-type into the p-type and creating an Ohmic contact.<sup>115</sup> A functionalized TMDC



MoS<sub>2</sub>(1-x)Se<sub>2x</sub>/graphene heterostructure with excellent quality, tunable structure, and electrical characteristics was recently reported by Yue *et al.* This heterostructure is crucial for phototransistors.<sup>116</sup> The electrical performance of monolayer MoS<sub>2</sub> can be enhanced by Sm doping. It is done by an atmospheric pressure chemical vapor deposition method. Sm ions introduced in monolayer MoS<sub>2</sub> led to an increase in the threshold voltage (from -12 to 0 V) for an Sm-doped MoS<sub>2</sub> field effect transistor. Furthermore, by doping of Sm, the electrical performance of the FET has been improved by 500%, with a 40% increase in mobility of the FET.<sup>117</sup> MXene nanoflake-doped WS<sub>2</sub> has a five-fold increase in PL efficiency, which is higher than that observed with electrical or plasma doping.<sup>118</sup> Ren-jie Chang *et al.* reported the post growth alteration of WS<sub>2</sub>, which was grown through chemical vapor deposition.<sup>119</sup> By employing a SnS precursor to introduce metallic Sn dopants, a variety of Sn/W/s composite 2D materials are produced. Fig. 9(a) reported by Shisheng Li *et al.* illustrates how substituting neighbouring elements (Re, V, or Nb) for the Mo and W atoms in group VI TMDCs can alter the electrical properties of the TMDCs.<sup>120</sup> This substitutional doping affects the band structure and electrical characteristics of 2D TMDCs. The methods for making mixed salt solutions for doping are depicted in Fig. 9(b), starting with the preparation of source precursors (Na<sub>2</sub>MoO<sub>4</sub>, Na<sub>2</sub>WO<sub>4</sub>, NaReO<sub>4</sub>, and NaVO<sub>3</sub>). These are used to produce mixed salt solutions with precise control over the doping concentration, such as XReMo, XVMo, XReW, and XVW. Fig. 9(c) provides details of the synthesis of Re- and V-doped TMDC monolayers. These doped monolayers are created by sulfurizing or selenizing the mixed salts using an ambient-pressure thermal chemical vapor deposition (CVD) process, following the spin-coating of the mixed salt solutions onto sapphire substrates. Fig. 9(d and e) shows typical transport curves illustrating the electrical behavior of Re- and V-doped MoS<sub>2</sub> and WS<sub>2</sub> monolayers. Fig. 9(f) schematically depicts Au or Pd contacts and includes an optical image of the device that may display the experimental setup. Fig. 9(g) compares the transport characteristics of WS<sub>2</sub> field-effect transistors (FETs) with those of Au, Pd, and V-doped WSe<sub>2</sub> to provide insight into the electrical performance of these materials. The projected density of states (PDOS) of WSe<sub>2</sub> monolayers is shown in Fig. 9(h), which shows details of the electronic structure of these doped materials.

## 5.2. Tuning by strain engineering

Applying strain engineering can also enhance the performance of self-powered photodetectors based on TMDCs.<sup>121</sup> TMDCs' optoelectronic characteristics may be deliberately strained to maximize self-powered photodetection. Tensile strain may be deliberately used to raise the bandgap of TMDCs in self-powered photodetectors, increasing their sensitivity to particular light wavelengths.<sup>122</sup> Higher photoresponsivity and spectrum selectivity can be attained as a result. Furthermore, by optimizing the strain, charge separation and transport mechanisms may be strengthened, increasing the photodetectors' self-sustainability and energy conversion efficiency.<sup>123</sup> In order to modify TMDCs' characteristics for self-powered photodetection, strain engineering is essential. This allows researchers

to create photodetectors that are not only light-sensitive but also capable of producing electricity from incident photons, which removes the requirement for an external power source. Strain in the crystal lattice modifies orbital interactions, coupling between neighboring atoms, and lattice constants to design the physical characteristics of a solid. The majority of early TMDC monolayers may have their electrical and magnetic characteristics changed by strain engineering, creating further opportunities for optoelectronic and spintronic applications down the road. Utilizing a CVD-grown technique for carrier mobility adjustment, strain control may be used to optimize 2D electronic devices. For group IVB TMDCs, the band gap rises with tensile strain; however, for ZrX<sub>2</sub> and HfX<sub>2</sub>, the band gap begins to drop at strain, ranging from 6–8%.<sup>124</sup> Group VB TMDCs can undergo ferromagnetism either because of tensile strain or because it enhances the existing ferromagnetism.<sup>124</sup> With the exception of CrTe<sub>2</sub>, TMDCs experience a direct to indirect band gap transition upon application of the tensile strain group VIB.<sup>125–127</sup> Frisenda's group investigates the short-term modulation of single-layer TMDCs' reflectance by biaxial strain. Calculations using density functional theory indicate that the electronic bandgap and other electrical characteristics of many early TMDCs are significantly influenced by the tensile strain. For group IV B TMDCs (TiX<sub>2</sub>, ZrX<sub>2</sub>, and HfX<sub>2</sub>), the increasing tensile strain causes the bandgap to grow, whereas for X = S, Se, the bandgap increases and begins to decrease.<sup>128</sup> It has already been demonstrated that atomic layered TMDCs can be engineered by using a variety of strain-engineering strategies, such as mechanical bending using a flexible substrate heated with a focused laser or applying a different force on a monolayer of MoS<sub>2</sub>.<sup>129</sup> MoS<sub>2</sub> exhibits a high Young's modulus and has a much higher mechanical strength than many high-tensile materials. This makes MoS<sub>2</sub> an ideal material for microelectronic applications.<sup>130</sup> MoS<sub>2</sub> is a promising microelectronic and straintronic material that can be manipulated by the application of strain to regulate its carrier density, effective mass, and mobility. Fig. 10(a) depicts the development of a flexible p–n photodetector by Junli Du *et al.* employing few-layer WSe<sub>2</sub> and ZnO nanobelts.<sup>131</sup> It provides a schematic representation of the steps involved in creating this photodetector as well as the arrangement of the materials and components. Fig. 10(b) displays an optical image of the flexible photodetector under tensile stress. An inset of this image displays a close-up of the photodetector. It provides the observer with a visual representation of the real device by displaying the electrodes, ZnO nanobelts, and few-layer WSe<sub>2</sub>. Fig. 10(c) provides a schematic illustration of the operation of the flexible p–n photodetector using the few-layer WSe<sub>2</sub>–ZnO nanobelt. It makes it easier to comprehend how the device transforms light into an electrical signal. Fig. 10(d) schematically displays the WSe<sub>2</sub>–ZnO photodetector installed on a flexible PET substrate along with the structural arrangement of the device. Fig. 10(e–i) displays the current–voltage (*I*–*V*) curves on both linear and logarithmic scales for the flexible WSe<sub>2</sub>–ZnO diode at 0% strain. These graphs give insight into the electrical behavior of the diode at various voltage levels as well as the *I*–*V*





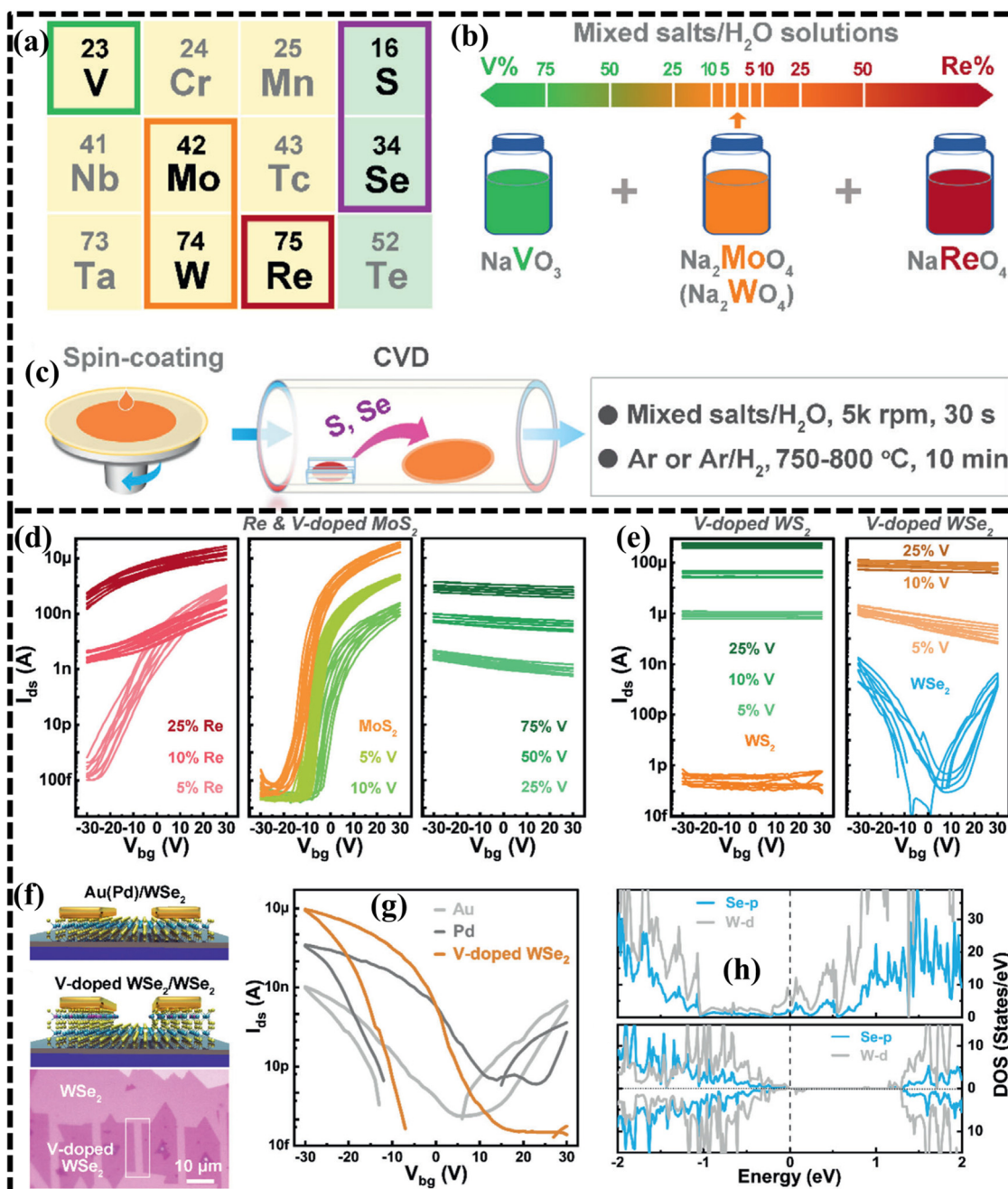


Fig. 9 (a) Partial periodic table showing substitutional Re and V doping to TMDCs, (b) mixed salt solutions with different  $\text{NaReO}_4$  and  $\text{NaVO}_3$ , (c) spin coating and the conditions employed in the CVD growth of Re and V doped TMDC monolayers, (d) and (e) typical transport curves of Re and V doped  $\text{MoS}_2$  and  $\text{WS}_2$  monolayers, (f) schematic of Au or Pd contacts and optical image of the device, (g) comparison of transport properties of  $\text{WS}_2$  FETs with Au, Pd and V-doped  $\text{WSe}_2$ , and (h) projected density of states (PDOS) of  $\text{WSe}_2$  monolayers. Reprinted with permission from ref. 120, Copyright of Wiley 2021.

characteristics of the  $\text{WSe}_2$ -ZnO photodetector at 0% strain under various optical light intensities.

### 5.3. Surface functionalization

One key tactic in creating self-powered photodetectors based on TMDCs is surface functionalization.<sup>132</sup> By changing their surface properties, such as  $\text{WSe}_2$  or  $\text{MoS}_2$ , TMDCs can perform better in self-powered photodetection applications. Surface

functionalization can be utilized to attach certain molecules or ligands, which will improve the TMDCs' interaction with incident light and boost their photoresponsivity and capacity to absorb light.<sup>133</sup> Enhancing charge separation and transport by functionalization can also result in more effective energy conversion. Furthermore, TMDC-based photodetectors may be made more reliable and stable by surface functionalization, which improves their suitability for real-world uses. All things



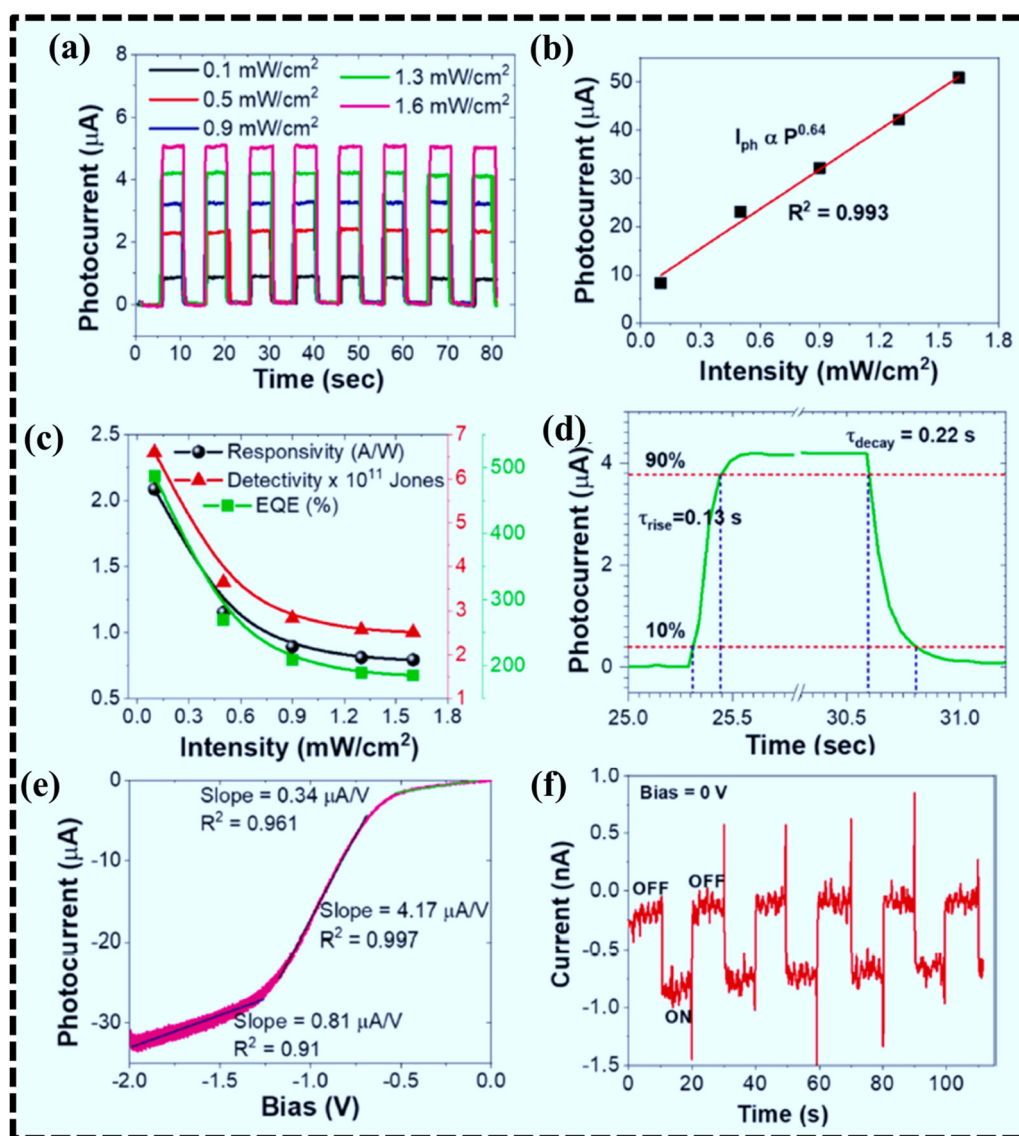
**Fig. 10** (a) Schematic fabrication process of the p-n photodetector with a WSe<sub>2</sub>-ZnO nanobelt, (b) optical image of the device under tensile strain, (c) device operation mechanism of the photodetector device, (d) schematic diagram of the device on the flexible PET substrate, (e) current-voltage characteristics under 0% strain, (f) current-voltage characteristics under different illumination intensities, (g) photovoltaic effect mechanism for self-powered photodetector device, and (h) and (i) photocurrent and responsivity with respect to the illumination intensities. Reprinted with permission from ref. 131 Copyright of Elsevier 2019.

considered, surface functionalization in TMDCs is essential for adjusting their surface characteristics to produce extremely sensitive, self-powered photodetectors for a variety of optoelectronic uses. In donor/acceptor interfaces, PCT occurs as a result of photon absorption. In this way, PCT is linked to all other excited-state decay processes and is therefore kinetically related

to their thermodynamics.<sup>134</sup> Monolayer TMDC semiconductors are characterized by excitonic transitions of the Brillouin zone *K* point, which dominate with their optical and optoelectronic properties.<sup>135,136</sup> TMDCs are capable of modulating excitonic transitions by altering their carrier concentrations, thereby controlling their optical properties.<sup>137</sup> According to

Jungwook *et al.* TMDCs can be modulated by chemical species adsorption. For example, when p-type dopants such as 2,3,5,6-tetrafluoro-7,7,8,8-tetracyanoquinodimethane (F4TCNQ) are interfaced with monolayer MoS<sub>2</sub> and WS<sub>2</sub>, excess electrons are decreased from these TMDCs such that their PL intensity increases by reducing negatively charged excitons and thus enhancing excitonic recombination.<sup>138,139</sup> When cesium carbonate (electron donating) is interfaced with monolayer MoS<sub>2</sub>, charge excitons are increased, such that PL intensity decreases.<sup>140</sup> An examination of the photodetection characteristics of Ag-WS<sub>2</sub>/Si heterojunction-based devices at various power intensities is presented by Meswa Patel *et al.*<sup>141</sup> The time-resolved photoresponse at a constant 2 V bias is shown in Fig. 11(a), which shows that current significantly increases

upon illumination and drops upon turning off the light. Exciton production at the Ag-WS<sub>2</sub>/Si interface is responsible for this. As light intensity rises, photocurrent increases, as may be seen in Fig. 11(b). The photoresponse is limited by carrier entrapment, as given by the relationship of  $I_{ph} \propto P^{0.64}$ . Fig. 11(c) shows the variations in detector parameters with light intensity. At 532 nm, 0.1 mW cm<sup>-2</sup> illumination, and -2 V bias, the maximum of responsivity, detectivity, and external quantum efficiency (EQE) are reached. The time-resolved photoresponse, with measured  $\tau_{rise}$  and  $\tau_{decay}$ , is displayed in Fig. 11(d). Because of biomolecular adsorption on defect sites,  $\tau_{decay}$  is somewhat greater, extending the life-span of the charge carrier. Under white light, photocurrent fluctuation with bias voltage is shown in Fig. 11(e), in which



**Fig. 11** (a) Time dependent photo response of the Ag-WS<sub>2</sub> photodetector device at different illumination intensities, (b) photocurrent with respect to the illumination intensities, (c) responsivity, detectivity and EQE of the photodetector device, (d) rise and decay time constant, (e) photocurrent with respect to the applied bias potential and (f) time dependent photo response under an applied bias potential of 0 V. Reprinted with permission from ref. 141, Copyright of Elsevier 2021.





photocurrent rises in three stages, linearly. The self-powered time-resolved photoresponse with notable photoresponsivity, detectivity, and EQE is displayed in Fig. 11(f) as a result of n-Ag-WS<sub>2</sub> and p-Si forming a type-II interface and junction. Furthermore, the Ag-WS<sub>2</sub>/Si spectral photoresponse is compared to WS<sub>2</sub>/Si heterojunctions, demonstrating higher photocurrent values for Ag-WS<sub>2</sub>/Si, particularly at 532 nm because of Ag nanoparticle surface plasmon resonances.

#### 5.4. Covalent functionalization

A particular technique for surface functionalization of TMDCs is called covalent functionalization, which entails the creation of robust covalent connections between the functional groups and the TMDC surface.<sup>142</sup> The optimization of TMDCs' performance in self-powered photodetection can be achieved by covalent functionalization. Covalent functionalization enables exact control over the insertion of certain molecules or chemical groups on the TMDC surface for self-powered photodetectors based on TMDCs. It is possible to modify these covalently bound groups to improve photoresponsivity, charge separation, and transport efficiency in addition to its light-absorbing capabilities.<sup>143</sup> In photodetection applications, covalent functionalization offers a stable and long-lasting alteration of the TMDC surface, guaranteeing long-term performance and dependability. TMDCs' optoelectronic characteristics may be precisely tuned to increase their sensitivity to incoming light and their ability to produce electrical energy from absorbed photons by carefully attaching the right functional groups through covalent bonds. This strategy may result in the creation of self-powered, incredibly effective photodetectors with specialized performance traits. Duo O. Li *et al.* demonstrated the direct covalent functionalization of semiconducting MoS<sub>2</sub>.<sup>144</sup> Mild chemistries are needed to allow for the direct covalent functionalization of semiconducting TMDC surfaces. It has been possible to covalently functionalize the basal plane of unaltered semiconducting MoS<sub>2</sub> using moderate conditions and aryl diazonium salts. This allows for the covalent binding of a range of functional groups to the surface of MoS<sub>2</sub>.<sup>145</sup>

#### 5.5. Non-covalent functionalization

An additional method for surface modification in TMDCs for self-powered photodetection is non-covalent functionalization.<sup>146</sup> In this way, specific molecules or functional groups are linked to the TMDC surface by weaker, non-covalent interactions such as hydrogen bonds, van der Waals forces, or electrostatic interactions.<sup>147</sup> Reversible changes and the preservation of TMDCs' inherent features are two benefits of non-covalent functionalization. Non-covalent functionalization enables accurate adjustment of the TMDC surface characteristics to improve its interaction with light in the context of self-powered photodetection. Adsorbed functional groups, such as organic compounds or polymers, can enhance the efficiency of charge separation, light absorption, and photoresponsivity on the TMDC surface. Increased photodetection sensitivity may result from this. Moreover, through charge transfer and other interactions at the interface, non-covalent functionalization

might make it easier to assemble layered structures or heterostructures using TMDCs and other materials, which may increase photodetection performance. The reason for non-covalent functionalization is appealing is that it preserves the inherent characteristics of TMDCs and may be reversed, meaning functional groups can be added or removed as needed. With this flexibility, the structural integrity and electrical characteristics of TMDCs may be preserved while optimizing their qualities for self-powered photodetection. Non covalent interactions play a crucial role in customizing the characteristics of nanomaterials. The initial exploration of this concept was carried out with graphene.<sup>148</sup> Subsequently, this approach was extended to inorganic counterparts, including black phosphorus and MoS<sub>2</sub>,<sup>149</sup> as the representative of TMDCs. Noncovalent functionalization of TMDCs has primarily been achieved on surfaces, with only a few exceptions.<sup>150</sup> Non-covalent molecular functionalization of materials revolves around properties arising from electrostatic forces, dipole-dipole interactions, induced dipole interactions, hydrogen bonding, and van der Waals forces. These methods of non-covalent interaction enable convenient surface modification without necessitating particular molecule-substrate bonds.<sup>151</sup> In recent years, the scientific community has directed its focus toward molecular methods that intricately adjust the characteristics of 2D materials. This is achieved by harnessing the extensive range of available molecular systems. The integration of these materials with photo responsive molecular systems offers a captivating and demanding avenue, holding the potential for revolutionary technologies in multifunctional optoelectronic applications.<sup>152</sup>

## 6. Device structures for self-powered photodetector devices

### 6.1. Phototransistor devices

When used in phototransistor mode, bipolar junction transistors (BJTs) may be set up to function as self-powered photodetectors. Fig. 12(a) shows the high responsive MoS<sub>2</sub> phototransistors based on charge trapping HfO<sub>2</sub> dielectrics and Fig. 12(b) shows the avalanche photo diode, whereas Fig. 12(c) shows the integrated MoS<sub>2</sub> phototransistors. Incident light causes electron-hole pairs to form in the base area, affecting the base current, while operating in the active zone, where the base-emitter junction is forward-biased. Consequently, this regulates the collector current, thereby increasing the photocurrent. Because they may give better gain and enhanced sensitivity when compared to solo photodiodes, phototransistors are highly valued for their intrinsic sensitivity to light. They are appropriate for applications needing quick light detection because of their comparatively quick reaction times. While BJTs do not produce energy on their own, they can be incorporated into systems that do, like photovoltaic cells or piezoelectric elements, to produce self-powered photodetectors that capture light and turn it into electrical signals for a variety of light-sensing uses. Bipolar junction transistors serve as the foundation for the type of photodetector known as a phototransistor (BJT). It has a light-sensitive area where the absorption





Fig. 12 (a) High responsivity in MoS<sub>2</sub> phototransistors based on charge trapping HfO<sub>2</sub> dielectrics,<sup>155</sup> (b) avalanche photo diode,<sup>156</sup> and (c) MoS<sub>2</sub> phototransistors. Reprinted with permission from ref. 157.

of light creates electron-hole pairs, which in turn modify the current passing through the transistor. It is made up of a p-n-p or n-p-n type semiconductor structure. An intrinsic (i-type) semiconductor, an n-type semiconductor, and a p-type semiconductor make up the three layers of a phototransistor, a form of photodetector based on a bipolar junction transistor (BJT). An alternative to this device is a phototransistor based on 2D transition metal dichalcogenides (TMDCs), which uses 2D TMDC materials as the semiconductor layers. The n-type layer of a phototransistor made of 2D TMDCs is often made of a 2D TMDC material doped with impurities like phosphorus or nitrogen, whereas the p-type layer is typically made of a 2D TMDC material doped with impurities like boron or aluminum.<sup>153</sup> Typically, the intrinsic layer is made of an impurity-doped 2D TMDC material. A light that enters the intrinsic layer of the phototransistor and is absorbed there produces electron-hole pairs that modify the current that passes through the component. Both the detection of light and the production of an electrical signal can be accomplished by modulating the current. The phototransistor based on 2D TMDCs offers many benefits, including high sensitivity, quick response times, and minimal noise. The device also has strong light-matter interaction and high electron mobility, which can improve its performance as a photodetector, thanks to the use of 2D TMDCs as the semiconductor layers. However, the phototransistor based on 2D TMDCs also has some constraints, such as the requirement to synthesize superior, homogenous 2D TMDC layers and the requirement to design the device structure and the doping. However, 2D TMDC-based phototransistors also have several drawbacks, including the necessity to construct high-quality, homogenous 2D TMDC layers as well as the need to design the structure of the device and doping levels to achieve good performance. These tools have been extensively employed in a wide range of applications, including sensing, imaging, and spectroscopy. They are also frequently utilized in the detection of electromagnetic radiation in the ultraviolet, visible, and infrared spectra as well as light-sensitive components in optical communication systems, such as fiber-optic networks.

Kaixi Shi *et al.* present a comprehensive overview of an InP@ZnS-MoS<sub>2</sub> hybrid energy transfer system and associated photodetector devices.<sup>154</sup> Fig. 13(a) offers a schematic representation of the energy transfer mechanism. Fig. 13(b) provides a structural diagram of the photodetector device, illustrating its physical layout. Fig. 13(c) and (d) show  $I_{ds}$ - $V_{ds}$  characteristics for the InP@ZnS-MoS<sub>2</sub> phototransistor and MoS<sub>2</sub> phototransistor, respectively, under various backgated voltages, revealing their current responses to different voltage conditions. Fig. 13(e) displays the gating response ( $I_{ds}$ - $V_{gs}$ ) for a fixed  $V_{ds}$ , showcasing how current changes with gate-source voltage. Fig. 13(f) highlights the phototransistors' photocurrent as a function of laser power density, while Fig. 13(g) demonstrates the photoresponsivity of the phototransistors under specific voltage and gate conditions as a function of laser power, offering insights into their light-detection capabilities and performance.

## 6.2. Heterostructured photodetectors

Innovative devices known as heterostructured self-powered photodetectors utilize the potential of mixing several semiconductor materials to enable improved light detection and self-sufficient functioning. These photodetectors have a thoughtfully constructed stack of several materials or layers, each selected for certain qualities. One of the semiconductor layers absorbs photons and produces electron-hole pairs when exposed to incoming light. Effective charge separation takes place at the interface between these materials, preventing these carriers from recombining. In order to convert the produced electron-hole pairs into electrical energy, the heterostructure can include energy-harvesting elements like solar cells or piezoelectric materials. Because of their clever design, which combines specialized materials and components for effective energy collecting and conversion, heterostructured devices are self-powered. Through the heterostructure combination of diverse semiconductor materials and energy conversion components such as thermoelectric modules, photovoltaic cells, or



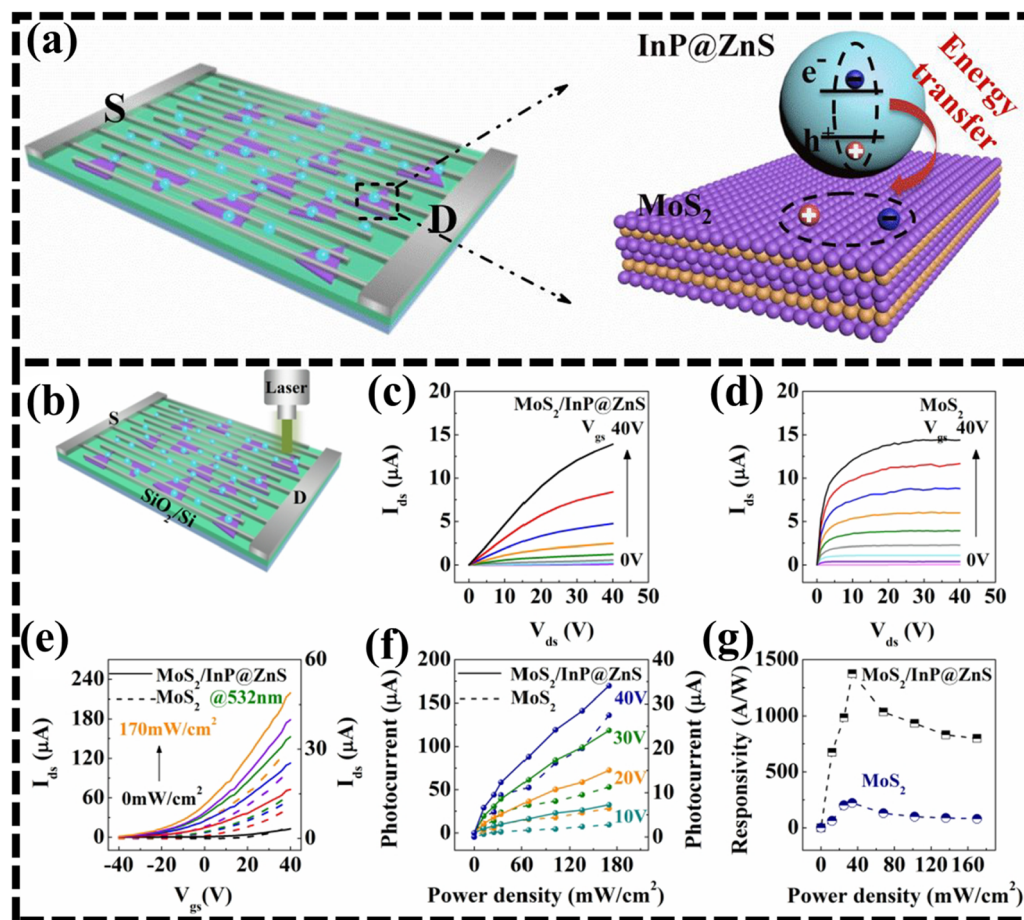


Fig. 13 (a) Schematic of the InP@ZnS-MoS<sub>2</sub> hybrid energy transfer system, (b) a structural diagram of the photodetector device, (c)  $I_{ds}$ - $V_{ds}$  characteristics under different backgated voltages for the InP@ZnS-MoS<sub>2</sub> phototransistor, (d)  $I_{ds}$ - $V_{ds}$  characteristics under different backgated voltages for the MoS<sub>2</sub> phototransistor, (e) gating response ( $I_{ds}$ - $V_{gs}$ ) for  $V_{ds} = 1$  V, (f) photocurrent of the phototransistors devices with laser power density and (g) photoresponsivity of the phototransistors at  $V_{ds} = 1$  V and  $V_{gs} = 40$  V as a function of laser power. Reprinted with permission from ref. 154, Copyright of American Chemical Society 2020.

piezoelectric components, these devices can independently convert a variety of energy sources, including mechanical stress, light, and temperature gradients, into electrical power. Electrical current is produced when created electron-hole pairs do not recombine because of the intrinsic charge separation at material surfaces. Self-powered devices in the semiconductor sector are included in the study of Wanhui Wu *et al.*<sup>158</sup> Epitaxially connected TMDCs' lateral heterojunctions show promise for self-powered devices due to their quick electron-hole separation. The photovoltaic light sensors based on the monolayer MoS<sub>2</sub>-WS<sub>2</sub> in-plane heterostructure exhibit a spectral responsivity ( $R_s$ ) of 4.36 mA W<sup>-1</sup> and a detectivity ( $D^*$ ) of  $4.36 \times 10^{13}$  Jones under 28.64 mW cm<sup>-2</sup> @532 nm. Because these devices use the inherent potential at the interface, they are interesting candidates for future applications in 2D heterostructures. A schematic representation of the MoS<sub>2</sub> and WS<sub>2</sub> heterostructures is shown in Fig. 14(a), and an optical picture of the apparatus is shown in Fig. 14(b). Fig. 14(d) shows the enlarged  $I$ - $V$  characteristics close to 0 V of applied bias, whereas Fig. 14(c) shows the logarithmic  $I$ - $V$  characteristics of

the device at various illumination intensities.  $I$ - $t$  characteristics for on-off switching are displayed in Fig. 14(e), and the outstanding self-powered photovoltaic mode at 0 V of applied bias potential is displayed in Fig. 14(f). A mixed-dimensional vertical heterostructure with a photovoltaic effect was created by Lixiang Han *et al.* by stacking 2D WS<sub>2</sub> nanosheets over 1D Te microwires; Fig. 14(g) shows the schematic of the heterostructure of that photodetector device.<sup>159</sup> This self-powered photodetector, when exposed to light at 635 nm, has exceptional photoresponsivity (471 mA W<sup>-1</sup>), external quantum efficiency (91%), and detectivity ( $1.24 \times 10^{12}$  Jones) with a maximum open-circuit voltage of around 0.2 V. The transfer curve of the device is shown in Fig. 14(h), and its output characteristics are shown in Fig. 14(i). The  $I$ - $V$  characteristics under both dark and light conditions are displayed in Fig. 14(j). Fig. 14(k) displays the photoresponsivity and net photocurrent in relation to illumination intensity, while Fig. 14(l) showcases its exceptional EQE and detectivity. The time-dependent photocurrent is shown in Fig. 14(m). It features low dark current (sub-pA level), fast reaction times, and a high  $I_{light}/I_{dark}$







Fig. 14 (a) Schematic diagram of the heterostructures of MoS<sub>2</sub> and WS<sub>2</sub>, (b) optical image of the device, (c) logarithmic  $I$ - $V$  characteristics of the device under different illumination intensities, (d) the magnified  $I$ - $V$  characteristics near 0 V of applied bias, (e)  $I$ - $t$  characteristics for on-off switching and (f) self-powered photovoltaic mode at 0 V of applied bias potential. Reprinted with permission from ref. 158, Copyright of Elsevier 2018. (g) Schematic diagram of the heterostructure photodetector device, (h) the transfer curve of the device, (i) output characteristics of the device, (j)  $I$ - $V$  characteristics under dark and illumination conditions, (k) photoresponsivity and the net photocurrent as a function of illumination intensity, (l) EQE and detectivity of the device, (m) time dependent photocurrent of the device, (n) time dependent photo response of the device and (o) stability of the photodetector device over 200 cycles. Reprinted with permission from ref. 159, Copyright of Royal Society Chemistry 2021.

ratio (104). These characteristics make 1D Te-2D WS<sub>2</sub> mixed-dimensional heterostructures promising for high-performance photovoltaics and photodetectors. The time-dependent

photoresponse is displayed in Fig. 14(n), and the outstanding stability of the photodetector after 200 cycles is verified in Fig. 14(o).



### 6.3. Metal–semiconductor–metal (MSM) and Schottky photodetectors

A photodetector known as a metal–semiconductor–metal (MSM) photodetector detects light using a metal–semiconductor–metal construction. A transition metal dichalcogenide, for example, often makes up the semiconductor layer in an MSM photodetector based on 2D materials (TMDC). The two metal layers, one on top and one at the bottom of the 2D material serve as electrodes to measure the photocurrent produced by light absorption. The Schottky contact between the semiconductor and the metal in an MSM photodetector creates an internal electric field that separates electron–hole pairs created by light absorption in the semiconductor layer. The schematic device construction of the photodetector with the MoS<sub>2</sub> active layer and Ti/Au electrodes is shown in Fig. 15(a), whereas Fig. 15(b) displays the *I*–*V* characteristics measured in the voltage range of  $-5$  to  $+5$  V.<sup>160</sup> Fig. 15(c) shows the responsivity of this device with different light intensities and Fig. 15(d) exhibits the responsivity with respect to applied bias at different illumination intensities. A current that can be measured as an output signal is produced as a result of the movement of electrons and holes. The advantages of MSM photodetectors based on 2D materials are high sensitivity, high responsivity, and quick response times. Moreover, the high electron mobility of the device and robust light–matter interactions, which can improve its performance as a photodetector, are made possible by the use of 2D TMDCs as the semiconductor layers.<sup>161</sup> The need for high-quality metal–semiconductor interfaces, as

well as the need to design the device structure and the doping levels to achieve good performance, are some of the limits of MSM photodetectors based on 2D materials. Spectroscopy, imaging, and sensing are just a few of the many applications that MSM photodetectors based on 2D materials are utilized in. They are also frequently utilized in the detection of electromagnetic radiation in the ultraviolet, visible, and infrared spectra as well as light-sensitive components in optical communication systems, such as fiber-optic networks. Moreover, a new generation of photomultipliers, which are extremely sensitive light detectors, has been developed using these technologies. MSM (metal–semiconductor–metal) based TMDC photodetectors are a class of photodetectors that use TMDCs as the active component. These photodetectors have a wide bandgap, high carrier mobility, and a high absorption coefficient, which provide them several benefits over conventional silicon-based photodetectors. The ability to detect light with extremely high sensitivity is one of the key benefits of TMDC-based MSM photodetectors. They are suitable for high-speed optical communication applications because of their quick response times, which are on the order of picoseconds. The ability to function at extremely low light levels, which makes them suited for low-light imaging applications, is another benefit of TMDC-based MSM photodetectors. The range of applications for these devices is also increased by the fact that they can function at a broad range of wavelengths, including in the ultraviolet and infrared sections of the spectrum. The need for high-quality TMDC materials and the necessity of adequate device



Fig. 15 (a) Lateral device schematic structure of the photodetector, (b) *I*–*V* characteristics measured in the voltage range of the photodetector, (c) responsivity of the device with different light intensities and (d) responsivity with respect to applied bias at different illumination intensities. Reprinted with permission of ref. 160, Copyright of Nature 2016.

construction and encapsulation to reduce environmental degradation are some of the difficulties faced by TMDC-based MSM photodetectors. Overall, TMDC-based MSM photodetectors are a promising technology with many potential uses, but more study and development are required to properly tap into their potential. Table 4 summarizes some functionalized TMDC MSM photodetectors with their important parameters. The absorption of light by the TMDC material, which results in the formation of electron-hole pairs, is the basis for the operation of TMDC-based MSM photodetectors. The internal electric field of the device then divides these electron-hole pairs, with the electrons going to the n-type contact and the holes to the p-type contact.<sup>162</sup> As a result, the device generates a current flow that may be monitored to find the existence of light. The TMDC material is commonly sandwiched between two metal contacts in MSM photodetectors based on TMDCs, thus the term “metal-semiconductor-metal” (MSM). In addition to acting as the device's electrical contacts, the metal contacts also combine with the TMDC material to form a Schottky barrier that aids in separating the electron-hole pairs and boosts the effectiveness. With TMDC-based MSM photodetectors, light can be absorbed through various methods, such as excitonic absorption, indirect inter-band transition, and direct inter-band transition. Depending on the type of TMDC materials utilized and the light wavelength being absorbed, a certain mechanism will occur. TMDC-based MSM photodetectors are renowned for their quick response times, which are on the order of picoseconds due to the high carrier mobility in TMDC materials, in addition to the aforementioned mechanism. The general operating principle of TMDC-based MSM

photodetectors is based on the light absorption by the TMDC material, which generates electron-hole pairs, and the separation of these charges by the integrated electric field, which yields a current flow that can be measured to detect the presence of light.<sup>163</sup> The signal produced by a photodetector, which in the case of TMDC-based MSM photodetectors is a current flow arising from the absorption of light, is amplified using the photomultiplication process. By increasing the signal-to-noise ratio of the detector, photomultiplication aims to make it more sensitive and capable of detecting weaker signals.<sup>164</sup> The employment of a cascade of amplifiers, each of which amplifies the signal before it is sent on to the next stage, is one method of achieving photomultiplication in TMDC-based MSM photodetectors. Trans-impedance amplification, the name given to this sort of photomultiplication, is achieved by cascading several trans-impedance amplifiers (TIAs). Using an approach known as “avalanche photodetection,” which accelerates the electron-hole pairs created by the absorption of light to crash with other atoms and produce additional electron-hole pairs, is another method for achieving photomultiplication in TMDC-based MSM photodetectors. Impact ionization has the capacity to increase the number of charge carriers and, consequently, the current flow, producing a stronger signal. Remember that to prevent noise and saturation, both procedures would need to be carefully planned. It is important to note that MSM photodetectors based on TMDCs have demonstrated great responsiveness and quick response times, making them appropriate for low-light imaging applications and high-speed optical communication applications even without the usage of photomultiplication. In some

**Table 4** Summary of the literature data for photodetectors based on two-dimensional transition metal dichalcogenides. The table contains some benchmark parameters (responsivity, EQE, and detectivity)

Materials	Device type	Bias/incident light ( $\text{V } \mu\text{W}^{-1} \text{ cm}^{-2}$ )	Responsivity ( $\text{A W}^{-1}$ )	Detectivity (Jones)	EQE (%)	Ref.
WSe <sub>2</sub> /PANI	MSM	15/1.0 × 10 <sup>4</sup>	0.056	5.8 × 10 <sup>9</sup>	—	165
GeSe/MoS <sub>2</sub>	Diode	0/3080	0.105	1.46 × 10 <sup>10</sup>	24.2	166
WS <sub>2</sub> /Al <sub>x</sub> /Ge	Diode	0/42	0.634	4.3 × 10 <sup>11</sup>	50	167
WS <sub>2</sub> /Si	Diode	0/16.5	0.290	2.6 × 10 <sup>14</sup>	—	168
WS <sub>2</sub> /GaAs	Diode	4/1.7 × 10 <sup>-2</sup>	0.527	1.03 × 10 <sup>14</sup>	—	169
WS <sub>2</sub> /Gr	MSM	—/10	8.050	2.8 × 10 <sup>10</sup>	1.8 × 10 <sup>3</sup>	170
Si/SiO <sub>2</sub> /WS <sub>2</sub>	Diode	0/0.27	2.31	9.16 × 10 <sup>11</sup>	22	171
p-WSe <sub>2</sub> /n-Si	Diode	0/—	0.67	1.59 × 10 <sup>13</sup>	85	172
SWCNT/MoS <sub>2</sub>	MSM	1/10	2.01 × 10 <sup>3</sup>	3.2 × 10 <sup>12</sup>	—	173
Bi <sub>2</sub> O <sub>3</sub> Se/MoTe <sub>2</sub>	Diode	60/—	4.96	3.84 × 10 <sup>12</sup>	7.21	174
MoS <sub>2</sub> (1-x)Se <sub>2x</sub>	Phototransistor	0/1.73 × 10 <sup>5</sup>	0.311	10 <sup>11</sup>	—	175
n-WS <sub>2</sub> /Si	Diode	—1/—	0.87	1.8 × 10 <sup>11</sup>	161	176
ZnO-WS <sub>2</sub> /Si	Diode	—2/200	9.48	10 <sup>12</sup>	1758	177
WSe <sub>2</sub> /Si	Diode	—	1.15	10 <sup>11</sup>	134	178
WSe <sub>2</sub> /MoS <sub>2</sub>	Phototransistor	20/5 × 10 <sup>-3</sup>	2700	5 × 10 <sup>11</sup>	—	179
WS <sub>2</sub> /Polyacrylamide	MSM	—/50	1.3742	1.52 × 10 <sup>12</sup>	465.93	52
Au-MoS <sub>2</sub> /Si	MSM	—/792	0.128	1.32 × 10 <sup>10</sup>	~1.2 × 10 <sup>8</sup>	180
WS <sub>2</sub>	MSM	—	1093.1	2.6 × 10 <sup>12</sup>	2.1 × 10 <sup>5</sup>	181
MoS <sub>2</sub> /i-Ge	MSM	—	0.7	—	56.1	182
WS <sub>2</sub> /AlO <sub>x</sub> /Ge	MSM	—/42	0.634	4.3 × 10 <sup>11</sup>	50.8	167
Graphene/WS <sub>2</sub> /Ge	MSM	0/0.86 × 10 <sup>5</sup>	7.55	3 × 10 <sup>12</sup>	—	183
MoS <sub>2</sub> -Gr-MoS <sub>2</sub>	Phototransistor	0.5/—	0.030	10 <sup>9</sup>	—	184
ZnO/WSe <sub>2</sub>	Phototransistor	—	0.541	4.23 × 10 <sup>9</sup>	—	185
ReS <sub>2</sub>	Phototransistor	—/0.47	10 <sup>6</sup>	10 <sup>13</sup>	—	186
GaTe	Phototransistor	5/0.29 × 10 <sup>3</sup>	274.3	10 <sup>12</sup>	—	187
Te/WS <sub>2</sub>	Phototransistor	0/—	0.033	1.57 × 10 <sup>6</sup>	—	188
MoS <sub>2</sub>	Phototransistor	0/—	10 <sup>4</sup>	10 <sup>13</sup>	—	189





circumstances, photomultiplication is another method that can be utilized to boost the sensitivity of the detector.

The performance of a self-powered photodetector depends on various parameters, including responsivity, detectivity, NEP, on-off ratio, LDR, EQE *etc.* These parameters are influenced by the properties of materials used. The photodetector's operation is based on three distinct stages: light harvesting, separation of photocarriers, and the subsequent transport of charges. The light absorption capacity significantly impacts the performance of optoelectronic devices. To attain optimal results, light capture technology has been developed to extend the effective path length of light propagation and enhance the efficiency of photon collection.<sup>190</sup> Following photon absorption, the generation of charge carriers is crucial, and the superior separation and collection of these carriers contribute to the improved performance of the photodetector. TMDCs may be doped to become either n-type (electron-donor) or p-type (hole-donor) by carefully adding certain impurities or foreign atoms. This will change their carrier concentration and produce an inherent potential. Because of its intrinsic potential, the photodetector can produce electricity on its own when light strikes it, making it a self-sufficient gadget that does not require an external power source.<sup>112</sup> By changing their surface properties, such as WSe<sub>2</sub> or MoS<sub>2</sub>, TMDCs can perform better in self-powered photodetection applications. Surface functionalization can be utilized to attach certain molecules or ligands, which will improve the TMDCs' interaction with incident light and boost their photoresponsivity and capacity to absorb light.<sup>191</sup> In photodetection applications, covalent functionalization offers a stable and long-lasting alteration of the TMDC surface, guaranteeing long-term performance and dependability. Tobias Scharl *et al.* investigated WS<sub>2</sub> through exfoliation and combined it with perylene diamides (C<sub>36</sub>H<sub>29</sub>NO), which absorb visible light and accept electrons, creating adaptable electron-donor acceptor hybrids.<sup>192</sup> Following the discovery of graphene, a variety of TMDCs have surfaced, showcasing remarkable performance utilized in photodetectors. However, despite their potential, TMDCs based photodetectors encounter numerous challenges. To achieve high performance and superior photodetectors, several approaches can be pursued: refining the preparation techniques for TMDCs to yield high quality outputs, and designing and optimizing antenna structures to align with detected wavelengths, thereby enhancing the photodetector sensitivity.

The significant possibility of TMDCs to transform the next generation of photodetectors, image sensors, optical transceivers, spectrometers, *etc.*, depends on the successful development of initial prototypes. The awareness of this potential depends on advancing both technology and manufacturing willingness levels, achieved through concerted efforts between researchers, semiconductor foundries, and key industrial companies. The introduction of novel materials, such as TMDCs, holds the promise of enhancing the efficiency of self-powered photodetectors through the improvement of the open circuit voltage. Sustained investigation of creative nanostructures that facilitate effective separation of photogenerated electron hole

pairs through simple and reliable fabrication methods remains a fundamental avenue for future research. With the introduction of high-performance semiconductor materials like TMDCs, there exists a significant opportunity to elevate performance of self-powered photodetectors by advancing both material design and device testing methodologies. Inspired by the popularity of wearable photodetectors in recent years, the next generation of photodetectors must be manufactured to represent greater adaptability and with improved intelligence. This naturally involves new criteria for the design, dimensions, and efficacy of both materials and structures.

## 7. Conclusions

With an emphasis on the potential of TMDCs to satisfy the market for high-performance photodetector devices, this analysis concludes with a thorough summary of the current status of self-powered photodetectors. Setting a systematic and beautiful scene, the introduction emphasized the significance of self-powered photodetectors in a range of applications and the ongoing research and innovation required in this area. The difficulties encountered by scientists in creating TMDC-based autonomous photodetectors were carefully investigated. These problems range from the synthesis and characterization of materials to the engineering and optimization of devices, underscoring the need for innovative and practical solutions in the years to come. The intrinsic qualities of TMDCs, which make them attractive options for self-powered photodetectors, were thoroughly examined in this study. High carrier mobilities, mechanical flexibility, thermal conductivity, strong spin-valley coupling, and advantageous optical and electronic properties are just a few of these attributes, which together highlight TMDCs' potential to meet the stringent requirements for self-powered photodetector applications. Furthermore, different functionalization strategies were covered in detail, including surface, covalent, and non-covalent functionalization, as well as avenues like doping, strain engineering, and photoinduced charge transfer for modulating optoelectronic properties. Taken together, these tactics offer a path forward for improving the efficiency of TMDC-based photodetectors. The use of TMDCs in self-powered photodetectors has great promise, and continued study and development in this area should result in ground-breaking discoveries that have an influence on a broad variety of applications, including communications, healthcare, and other fields. Exciting opportunities exist for the creation and application of effective, environmentally friendly, and multipurpose photodetector devices in the future. Currently, enhancing information exchange between humans and technology is crucial, particularly in fields such as communication, healthcare, and environmental monitoring. The demand for improved electronic products has grown, emphasizing the necessity for self-powered devices with straightforward structures; these devices offer practical advantages, including minimal maintenance requirements, low or no external power, wireless functionality, self-sustainability, and a prolonged



lifespan, aiming to reduce complications and dependencies. The absorption of weak optical signals is delayed by the photogenerated carriers, necessitating an additional mechanism to enhance both the depletion region width and the electric field at the P–N junction within the self-powered photodetector. Consequently, addressing these challenges in self-powered photodetectors involves TMDCs under investigation.

## Author contributions

Alka Rani: investigation and writing – original draft, Arpit Verma: investigation, review, and editing, and B. C. Yadav: validation and supervision.

## Conflicts of interest

The authors declare that they have no known competing financial interest or personal relationships that could have appeared to influence the work reported in this paper.

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## References

- W. Ahmad, J. Wu, Q. Zhuang, A. Neogi and Z. Wang, *Small*, 2023, **19**, 2207641.
- F. H. L. Koppens, T. Mueller, P. Avouris, A. C. Ferrari, M. S. Vitiello and M. Polini, *Nat. Nanotechnol.*, 2014, **9**, 780–793.
- A. Verma, P. Chaudhary, R. K. Tripathi and B. C. Yadav, *Sustainable Energy Fuels*, 2021, **5**, 1394–1405.
- X. Bao, Q. Ou, Z. Q. Xu, Y. Zhang, Q. Bao and H. J. A. M. T. Zhang, *Adv. Mater. Technol.*, 2018, **3**, 1800072.
- K. Kalantar-zadeh, J. Z. Ou, T. Daeneke, M. S. Strano, M. Pumera and S. L. Gras, *Adv. Funct. Mater.*, 2015, **25**, 5086–5099.
- J. Ping, Z. Fan, M. Sindoro, Y. Ying and H. Zhang, *Adv. Funct. Mater.*, 2017, **27**, 1605817.
- Q. Liang, Q. Zhang, X. Zhao, M. Liu and A. T. S. Wee, *ACS Nano*, 2021, **15**, 2165–2181.
- M. Chhowalla, H. S. Shin, G. Eda, L.-J. Li, K. P. Loh and H. J. Zhang, *Nat. Chem.*, 2013, **5**, 263–275.
- L. Dou, Y. Yang, J. You, Z. Hong, W.-H. Chang, G. Li and Y. Yang, *Nat. Commun.*, 2014, **5**, 5404.
- C. X. Li, C. Chen, L. Zhao and N. Ma, *ACS Appl. Mater. Interfaces*, 2023, **15**, 23402–23411.
- X. Xue, C. Ling, H. Ji, J. Wang, C. Wang, H. Lu and W. Liu, *ACS Appl. Mater. Interfaces*, 2023, **15**, 5411–5419.
- C. K. Zankat, P. M. Pataniya, A. Patel, S. A. Bhakhar, S. Narayan, G. K. Solanki, K. D. Patel, V. M. Pathak, C. K. Sumesh and P. K. Jha, *Mater. Today Energy*, 2020, **18**, 100550.
- J. Zhong, X. Zhang and G. Xiang, *ACS Appl. Mater. Interfaces*, 2023, **15**, 35753–35787.
- M. Sun, P. Yang, D. Xie, Y. Sun, J. Xu, T. Ren and Y. Zhang, *Adv. Electron. Mater.*, 2019, **5**, 1800580.
- L. H. Zeng, S. H. Lin, Z. J. Li, Z. X. Zhang, T. F. Zhang, C. Xie, C. H. Mak, Y. Chai, S. P. Lau and L. B. Luo, *Adv. Funct. Mater.*, 2018, **28**, 1705970.
- K. Murali and K. Majumdar, *IEEE Trans. Electron Devices*, 2018, **65**, 4141–4148.
- D. B. Velusamy, R. H. Kim, S. Cha, J. Huh, R. Khazaeinezhad, S. H. Kassani, G. Song, S. M. Cho, S. H. Cho and I. Hwang, *Nat. Commun.*, 2015, **6**, 8063.
- C. B. Vining, *J. Appl. Phys.*, 1991, **69**, 331–341.
- S. Manzeli, D. Ovchinnikov, D. Pasquier, O. V. Yazyev and A. Kis, *Nat. Rev. Mater.*, 2017, **2**, 1–15.
- S. H. Mir, V. K. Yadav and J. K. Singh, *ACS Omega*, 2020, **5**, 14203–14211.
- F. Yan, Z. Wei, X. Wei, Q. Lv, W. Zhu and K. Wang, *Small Methods*, 2018, **2**, 1700349.
- W. Choi, N. Choudhary, G. H. Han, J. Park, D. Akinwande and Y. H. Lee, *Mater. Today*, 2017, **20**, 116–130.
- Y. Fang, Y. Ge, C. Wang and H. Zhang, *Laser Photonics Rev.*, 2020, **14**, 1900098.
- Z. Huang, W. Zhang and W. J. Zhang, *Materials*, 2016, **9**, 716.
- Z. Ye, T. Cao, K. O'Brien, H. Zhu, X. Yin, Y. Wang, S. G. Louie and X. Zhang, *J. Nat.*, 2014, **513**, 214–218.
- W. Choi, J. Ahn, K. T. Kim, H. J. Jin, S. Hong, D. K. Hwang and S. Im, *Adv. Mater.*, 2021, **33**, 2103079.
- L. Huang, A. Krasnok, A. Alú, Y. Yu, D. Neshev and A. E. Miroshnichenko, *Rep. Prog. Phys.*, 2022, **85**, 046401.
- A. Verma and B. C. Yadav, *ACS Appl. Nano Mater.*, 2023, **6**, 5493–5507.
- M. Bernardi, M. Palummo and J. C. J. Grossman, *Nano Lett.*, 2013, **13**, 3664–3670.
- O. Lopez-Sanchez, D. Lembke, M. Kayci, A. Radenovic and A. J. Kis, *Nat. Nanotechnol.*, 2013, **8**, 497–501.
- J. Cheng, C. Wang, X. Zou and L. J. Liao, *Adv. Opt. Mater.*, 2019, **7**, 1800441.
- A. Autere, H. Jussila, Y. Dai, Y. Wang, H. Lipsanen and Z. Sun, *Adv. Mater.*, 2018, **30**, 1705963.
- A. Manikandan, Y.-Z. Chen, C.-C. Shen, C.-W. Sher, H.-C. Kuo and Y.-L. Chueh, *Prog. Quantum Electron.*, 2019, **68**, 100226.
- W. Yu, S. Li, Y. Zhang, W. Ma, T. Sun, J. Yuan, K. Fu and Q. Bao, *J. Small*, 2017, **13**, 1700268.
- P. Pataniya, C. K. Zankat, M. Tannarana, C. K. Sumesh, S. Narayan, G. K. Solanki, K. D. Patel, V. M. Pathak and P. K. Jha, *ACS Appl. Nano Mater.*, 2019, **2**, 2758–2766.
- H. Kumar and S. Jit, *Handbook of II–VI Semiconductor-Based Sensors and Radiation Detectors: Volume 2*, Photodetectors, Springer, 2023, pp. 495–515.
- S. Ahmed and J. Yi, *Nano-Micro Lett.*, 2017, **9**, 1–23.
- C. Mu, J. Xiang and Z. Liu, *J. Mater. Res.*, 2017, **32**, 4115–4131.



- 39 A. Singh, A. Verma, B. C. Yadav and P. Chauhan, *Dalton Trans.*, 2022, **51**, 7864–7877.
- 40 A. Verma, P. Chaudhary, R. K. Tripathi, A. Singh and B. C. Yadav, *J. Inorg. Organomet. Polym. Mater.*, 2022, **32**, 2807–2826.
- 41 T. Ouyang, X. Zhao, X. Xun, F. Gao, B. Zhao, S. Bi, Q. Li, Q. Liao and Y. Zhang, *Adv. Sci.*, 2023, 2301585.
- 42 M. Reddeppa, S. B. Mitta, T. Chandrakalavathi, B.-G. Park, G. Murali, R. Jeyalakshmi, S.-G. Kim, S. H. Park and M.-D. Kim, *Curr. Appl. Phys.*, 2019, **19**, 938–945.
- 43 R. Raj, P. Lohia, D. Dwivedi, A. Verma and B. C. Yadav, *J. Mater. Sci.: Mater. Electron.*, 2022, **33**, 17939–17948.
- 44 R. Chai, Z. Lou and G. Shen, *J. Mater. Chem. C*, 2019, **7**, 4581–4586.
- 45 M. Kumar, J. Lim, H. Kang, S. Kim and H. Seo, *Nano Energy*, 2021, **82**, 105668.
- 46 W. Tian, Y. Wang, L. Chen and L. Li, *Small*, 2017, **13**, 1701848.
- 47 T. Lulé, S. Benthien, H. Keller, F. A. M. F. Mutze, P. Rieve, K. Seibel, M. Sommer and M. A. B. M. Bohm, *IEEE Trans. Electron Devices*, 2000, **47**, 2110–2122.
- 48 L. Su, W. Yang, J. Cai, H. Chen and X. Fang, *Small*, 2017, **13**, 1701687.
- 49 M. Gao, L. Yu, Q. Lv, F. Kang, Z.-H. Huang and R. Lv, *J. Materiomics*, 2023, **9**, 768–786.
- 50 K. Han, G. Huang, Y. Jia, Q. Niu, Z. Zheng and B. Wang, *Opt. Mater.*, 2024, **148**, 114971.
- 51 S. Maity, K. Sarkar and P. Kumar, *J. Mater. Chem. C*, 2021, **9**, 14532–14572.
- 52 A. Verma, P. Chaudhary, R. K. Tripathi and B. C. Yadav, *Mater. Adv.*, 2022, **3**, 3994–4005.
- 53 P. Stallinga, *Electrical characterization of organic electronic materials and devices*, John Wiley & Sons, 2009.
- 54 F. Giubileo and A. Di Bartolomeo, *Prog. Surf. Sci.*, 2017, **92**, 143–175.
- 55 J. Wang, Q. Yao, C. W. Huang, X. Zou, L. Liao, S. Chen, Z. Fan, K. Zhang, W. Wu and X. Xiao, *Adv. Mater.*, 2016, **28**, 8302–8308.
- 56 M. Kim, K. Y. Ma, H. Kim, Y. Lee, J. H. Park and H. S. Shin, *Adv. Mater.*, 2022, 2205520.
- 57 A. Sebastian, R. Pendurthi, T. H. Choudhury, J. M. Redwing and S. Das, *Nat. Commun.*, 2021, **12**, 693.
- 58 N. A. Jayah, H. Yahaya, M. R. Mahmood, T. Terasako, K. Yasui and A. M. Hashim, *Nanoscale Res. Lett.*, 2015, **10**, 1–10.
- 59 R. K. Willardson and A. C. Beer, *Semiconductors and semimetals*, Academic Press, 1977.
- 60 N. Ma, N. Tanen, A. Verma, Z. Guo, T. Luo, H. G. Xing and D. Jena, *Appl. Phys. Lett.*, 2016, **109**, 212101.
- 61 P. Bagheri, C. Quiñones-Garcia, D. Khachariya, S. Rathkanthiwar, P. Reddy, R. Kirste, S. Mita, J. Tweedie, R. Collazo and Z. Sitar, *J. Appl. Phys.*, 2022, **132**, 185703.
- 62 C. G. Rodrigues, *Microelectron. J.*, 2006, **37**, 657–660.
- 63 M. Nathan, W. Dumke, K. Wrenner, S. Tiwari, S. Wright and K. Jenkins, *Appl. Phys. Lett.*, 1988, **52**, 654–656.
- 64 S. Kim, A. Konar, W.-S. Hwang, J. H. Lee, J. Lee, J. Yang, C. Jung, H. Kim, J.-B. Yoo and J.-Y. Choi, *Nat. Commun.*, 2012, **3**, 1011.
- 65 H. Wang, C. H. Chan, C. H. Suen, S. P. Lau and J.-Y. Dai, *ACS Nano*, 2019, **13**, 6008–6016.
- 66 X.-Y. Gao, J.-M. Zhang, A. Ali, X.-M. Wei and Y.-H. Huang, *Thin Solid Films*, 2021, **732**, 138790.
- 67 M. Hamada, K. Matsuura, T. Sakamoto, I. Muneta, T. Hoshii, K. Kakushima, K. Tsutsui and H. Wakabayashi, *IEEE J. Electron Devices Soc.*, 2019, **7**, 1258–1263.
- 68 T. Kanazawa, T. Amemiya, A. Ishikawa, V. Upadhyaya, K. Tsuruta, T. Tanaka and Y. Miyamoto, *Sci. Rep.*, 2016, **6**, 22277.
- 69 I. Setiyawati, K.-R. Chiang, H.-M. Ho and Y.-H. Tang, *Chin. J. Phys.*, 2019, **62**, 151–160.
- 70 Y. Ma, A. Kuc, Y. Jing, P. Philipsen and T. Heine, *Angew. Chem., Int. Ed.*, 2017, **56**, 10214–10218.
- 71 H. Chen, Z. Li, L. Guo and X. Chen, *Europhys. Lett.*, 2017, **117**, 27009.
- 72 H. C. Movva, A. Rai, S. Kang, K. Kim, B. Fallahazad, T. Taniguchi, K. Watanabe, E. Tutuc and S. K. Banerjee, *ACS Nano*, 2015, **9**, 10402–10410.
- 73 T. Li, Z. Zhang, W. Zheng, Y. Lv and F. Huang, *AIP Adv.*, 2016, **6**, 115207.
- 74 Y. Wang, T. Sohler, K. Watanabe, T. Taniguchi, M. J. Verstraete and E. Tutuc, *Appl. Phys. Lett.*, 2021, **118**, 102105.
- 75 C. Xie and F. Yan, *Small*, 2017, **13**, 1701822.
- 76 P. Wang, Y. Lan, C. Huan, J. Luo, W. Cai, J. Fan, X. He, Z. Huang, L. Zhu and Q. Ke, *Mater. Sci. Eng., R*, 2023, **156**, 100759.
- 77 P. Chauhan, A. B. Patel, G. K. Solanki, H. K. Machhi, S. S. Soni, V. M. Pathak, V. Patel, S. Narayan and P. K. Jha, *Adv. Opt. Mater.*, 2021, **9**, 2100993.
- 78 A. Verma, A. Singh, P. Chaudhary, R. K. Tripathi, B. C. Yadav, P. Chauhan and D. Kumar, *Mater. Adv.*, 2023, **4**, 1062–1074.
- 79 H. Wang, H. Yuan, S. S. Hong, Y. Li and Y. Cui, *Chem. Soc. Rev.*, 2015, **44**, 2664–2680.
- 80 M. Tebyetekerwa, J. Zhang, Z. Xu, T. N. Truong, Z. Yin, Y. Lu, S. Ramakrishna, D. Macdonald and H. T. Nguyen, *ACS Nano*, 2020, **14**, 14579–14604.
- 81 A. Michail, J. Parthenios, D. Anastopoulos, C. Galiotis, M. Christian, L. Ortolani, V. Morandi and K. Papagelis, *2D Mater.*, 2018, **5**, 035035.
- 82 Y. Zou, Z. Zhang, J. Yan, L. Lin, G. Huang, Y. Tan, Z. You and P. Li, *Nat. Commun.*, 2022, **13**, 4372.
- 83 L. Peng, Y. Zhu, H. Li and G. Yu, *Small*, 2016, **12**, 6183–6199.
- 84 L. Lv, H. Ai, T. Chen, W. Zhu, Y. Guo, L. Dong and S. Song, *J. Mater. Chem. A*, 2023, **11**, 7115–7127.
- 85 B. Wang, Y. Yan, B. Qin, Z. Ye, J. Cao and J. Qi, *Chem. Eng. J.*, 2023, **476**, 146535.
- 86 X. Gu and C. Y. Zhao, *Comput. Mater. Sci.*, 2019, **165**, 74–81.
- 87 M. H. Khan, H. K. Liu, X. Sun, Y. Yamauchi, Y. Bando, D. Golberg and Z. Huang, *Mater. Today*, 2017, **20**, 611–628.
- 88 D. Akinwande, N. Petrone and J. Hone, *Nat. Commun.*, 2014, **5**, 5678.
- 89 G. Ding, G. Y. Gao, Z. Huang, W. Zhang and K. Yao, *Nanotechnology*, 2016, **27**, 375703.





- 90 A. Ansh, U. Patbhaje, J. Kumar, A. Meersha and M. Shrivastava, *Commun. Mater.*, 2023, **4**, 8.
- 91 W.-X. Zhou and K.-Q. Chen, *Sci. Rep.*, 2015, **5**, 15070.
- 92 X. Hu, L. Yan, L. Ding, N. Zheng, D. Li, T. Ji, N. Chen and J. Hu, *Coord. Chem. Rev.*, 2024, **499**, 215504.
- 93 C. S. Lau, S. Das, I. A. Verzhbitskiy, D. Huang, Y. Zhang, T. Talha-Dean, W. Fu, D. Venkatakrishnarao and K. E. Johnson Goh, *ACS Nano*, 2023, **17**, 9870–9905.
- 94 Y. Zhang, J. van den Brink, C. Felser and B. Yan, *2D Mater.*, 2018, **5**, 044001.
- 95 S. Yang, Y. Chen and C. Jiang, *InfoMat*, 2021, **3**, 397–420.
- 96 S. Chen, Y. Fu, M. Ishaq, C. Li, D. Ren, Z. Su, X. Qiao, P. Fan, G. Liang and J. Tang, *InfoMat*, 2023, e12400.
- 97 Z. Li, T. Yan and X. Fang, *Nat. Rev. Mater.*, 2023, **8**, 587–603.
- 98 L. Liao, E. Kovalska, J. Regner, Q. Song and Z. Sofer, *Small*, 2023, 2303638.
- 99 M. van der Laan, E. Heemskerk, F. Kienhuis, N. Diepeveen, D. Poonia, S. Kinge, M. T. Dang, V. A. Dinh, L. D. A. Siebbeles and A. Isaeva, *ACS Photonics*, 2023, **10**, 3115–3123.
- 100 M. R. Molas, K. Gołasa, Ł. Bala, K. Nogajewski, M. Bartos, M. Potemski and A. Babiński, *Sci. Rep.*, 2019, **9**, 1989.
- 101 V. Iberi, L. Liang, A. V. Ievlev, M. G. Stanford, M.-W. Lin, X. Li, M. Mahjouri-Samani, S. Jesse, B. G. Sumpter, S. V. Kalinin, D. C. Joy, K. Xiao, A. Belianinov and O. S. Ovchinnikova, *Sci. Rep.*, 2016, **6**, 30481.
- 102 H.-L. Liu, T. Yang, J.-H. Chen, H.-W. Chen, H. Guo, R. Saito, M.-Y. Li and L.-J. Li, *Sci. Rep.*, 2020, **10**, 15282.
- 103 T. N. Lin, S. Santiago, S. P. Caigas, C. T. Yuan, T. Y. Lin, J. L. Shen and Y. F. Chen, *npj 2D Mater. Appl.*, 2019, **3**, 46.
- 104 R. Dong and I. Kuljanishvili, *J. Vac. Sci. Technol., B: Nanotechnol. Microelectron.: Mater., Process., Meas., Phenom.*, 2017, **35**, 030803.
- 105 S. Sahare, P. Ghoderao, M. K. Sharma, M. Solovan, R. Aepuru, M. Kumar, Y. Chan, Z. Ziólek, S.-L. Lee and Z.-H. Lin, *Nano Energy*, 2023, **107**, 108172.
- 106 I. Zulfikar, M. A. Khan, S. Gul, N. U. L. Hassan, M. A. Rehman, M. A. Basit, H. W. Khalil, M. Ouladsmame, S. Rehman and M. F. Khan, *Phys. B*, 2023, **669**, 415313.
- 107 C. Gautam, A. Verma, P. Chaudhary and B. C. Yadav, *Opt. Mater.*, 2022, **123**, 111860.
- 108 X.-G. Zhao, Z. Shi, X. Wang, H. Zou, Y. Fu and L. Zhang, *InfoMat*, 2021, **3**, 201–211.
- 109 T. Kim, J. Lim, J. Byeon, Y. Cho, W. Kim, J. Hong, S. Jin Heo, J. Eun Jang, B.-S. Kim, J. Hong, S. Pak and S. Cha, *Small Struct.*, 2023, **4**, 2200274.
- 110 K. Novoselov, *J. Sci.*, 2016, **353**, 6298.
- 111 P. Zhao, D. Kiriya, A. Azcatl, C. Zhang, M. Tosun, Y.-S. Liu, M. Hettick, J. S. Kang, S. McDonnell and S. J. KC, *ACS Nano*, 2014, **8**, 10808–10814.
- 112 S. Wang, D. Ding, P. Li, Y. Sui, G. Liu, S. Zhao, R. Xiao, C. Tian, Z. Chen and H. Wang, *Small*, 2023, 2301027.
- 113 S. Mubeen, T. Zhang, B. Yoo, M. A. Deshusses and N. V. Myung, *J. Phys. Chem. C*, 2007, **111**, 6321–6327.
- 114 J. L. Johnson, A. Behnam, S. J. Pearton and A. Ural, *Adv. Mater.*, 2010, **22**, 4877–4880.
- 115 W. Hu, T. Wang, R. Zhang and J. Yang, *J. Phys. Chem. C*, 2016, **4**, 1776–1781.
- 116 Y. Yue, Y. Feng, J. Chen, D. Zhang and W. J. Feng, *J. Mater. Chem. C*, 2017, **5**, 5887–5896.
- 117 S. Li, S. Tian, Y. Yao, M. He, L. Chen, Y. Zhang and J. Zhai, *Nanomaterials*, 2021, **11**, 769.
- 118 Y. Tao, S. W. Koh, X. Yu, C. Wang, H. Liang, Y. Zhang, H. Li and Q. J. Wang, *Nanoscale Adv.*, 2019, **1**, 4783–4789.
- 119 R.-J. Chang, Y. Sheng, G. H. Ryu, N. Mkhize, T. Chen, Y. Lu, J. Chen, J. K. Lee, H. Bhaskaran and J. H. Warner, *ACS Appl. Mater. Interfaces*, 2019, **11**, 24279–24288.
- 120 S. Li, J. Hong, B. Gao, Y. C. Lin, H. E. Lim, X. Lu, J. Wu, S. Liu, Y. Tateyama and Y. Sakuma, *Adv. Sci.*, 2021, **8**, 2004438.
- 121 D. Bhattacharya, S. Mukherjee, R. K. Mitra and S. K. Ray, *Nanotechnology*, 2023, **34**, 435401.
- 122 K. Thakar and S. Lodha, *2D Materials for Electronics, Sensors and Devices*, Elsevier, 2023, pp. 207–258.
- 123 C. S. Rout, *2D Materials-Based Electrochemical Sensors*, Elsevier, 2023.
- 124 H. Guo, N. Lu, L. Wang, X. Wu and X. C. Zeng, *J. Phys. Chem. C*, 2014, **118**, 7242–7249.
- 125 P. Johari and V. B. J. A. n Shenoy, *ACS Nano*, 2012, **6**, 5449–5456.
- 126 W. S. Yun, S. Han, S. C. Hong, I. G. Kim and J. J. Lee, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2012, **85**, 033305.
- 127 Q. Yue, J. Kang, Z. Shao, X. Zhang, S. Chang, G. Wang, S. Qin and J. J. Li, *Phys. Lett. A*, 2012, **376**, 1166–1170.
- 128 H. Guo, N. Lu, L. Wang, X. Wu and X. C. Zeng, *J. Phys. Chem. C*, 2014, **118**, 7242–7249.
- 129 Y. Chen, W. Deng, X. Chen, Y. Wu, J. Shi, J. Zheng, F. Chu, B. Liu, B. An and C. You, *Nano Res.*, 2021, 1–7.
- 130 J. Li, Z. Shan and E. Ma, *MRS Bull.*, 2014, **39**, 108–114.
- 131 J. Du, Q. Liao, M. Hong, B. Liu, X. Zhang, H. Yu, J. Xiao, L. Gao, F. Gao and Z. Kang, *Nano Energy*, 2019, **58**, 85–93.
- 132 M. A. K. Purbayanto, M. Chandel, M. Birowska, A. Rosenkranz and A. M. Jastrzębska, *Adv. Mater.*, 2023, 2301850.
- 133 S. Roy, Aastha, K. A. Deo, K. Dey, A. K. Gaharwar and A. Jaiswal, *ACS Appl. Mater. Interfaces*, 2023, **15**, 35753–35787.
- 134 D. B. Sulas-Kern, E. M. Miller and J. L. Blackburn, *Energy Environ. Sci.*, 2020, **13**, 2684–2740.
- 135 K. F. Mak, C. Lee, J. Hone, J. Shan and T. F. J. Heinz, *Phys. Rev. Lett.*, 2010, **105**, 136805.
- 136 W. Zhao, Z. Ghorannevis, L. Chu, M. Toh, C. Kloc, P.-H. Tan and G. J. Eda, *ACS Nano*, 2013, **7**, 791–797.
- 137 J. Choi, H. Zhang and J. H. J. Choi, *ACS Nano*, 2016, **10**, 1671–1680.
- 138 S. Mouri, Y. Miyauchi and K. J. Matsuda, *Nano Lett.*, 2013, **13**, 5944–5948.
- 139 N. Peimyoo, W. Yang, J. Shang, X. Shen, Y. Wang and T. J. Yu, *ACS Nano*, 2014, **8**, 11320–11329.
- 140 J. D. Lin, C. Han, F. Wang, R. Wang, D. Xiang, S. Qin, X.-A. Zhang, L. Wang, H. Zhang and A. T. S. J. Wee, *ACS Nano*, 2014, **8**, 5323–5329.



- 141 M. Patel, P. M. Pataniya, D. J. Late and C. K. Sumesh, *Appl. Surf. Sci.*, 2021, **538**, 148121.
- 142 B. Chen, S. Sui, F. He, C. He, H.-M. Cheng, S.-Z. Qiao, W. Hu and N. Zhao, *Chem. Soc. Rev.*, 2023, 7802–7847.
- 143 B. Han, S. M. Gali, S. Dai, D. Beljonne and P. Samorì, *ACS Nano*, 2023, **17**, 17956–17965.
- 144 D. O. Li, M. S. Gilliam, X. S. Chu, A. Yousaf, Y. Guo, A. A. Green and Q. H. Wang, *Mol. Syst. Des. Eng.*, 2019, **4**, 962–973.
- 145 X. S. Chu, A. Yousaf, D. O. Li, A. A. Tang, A. Debnath, D. Ma, A. A. Green, E. J. Santos and Q. H. Wang, *Chem. Mater.*, 2018, **30**, 2112–2128.
- 146 X. Liu, Y. Niu, D. Jin, J. Zeng, W. Li, L. Wang, Z. Hou, Y. Feng, H. Li and H. Yang, *J. Colloid Interface Sci.*, 2023, **649**, 909–917.
- 147 A. S. Siddiqui, A. Hayat, L. A. G. Pohan, M. A. Ahmad and M. Nasir, *Mater. Today Chem.*, 2023, **32**, 101625.
- 148 N. V. Kozhemyakina, J. M. Englert, G. Yang, E. Spiecker, C. D. Schmidt, F. Hauke and A. Hirsch, *Adv. Mater.*, 2010, **22**, 5483–5487.
- 149 A. J. Molina-Mendoza, L. Vaquero-Garzon, S. Leret, L. de Juan-Fernández, E. M. Pérez and A. Castellanos-Gomez, *Chem. Commun.*, 2016, **52**, 14365–14368.
- 150 E. P. Nguyen, B. J. Carey, C. J. Harrison, P. Atkin, K. J. Berean, E. Della Gaspera, J. Z. Ou, R. B. Kaner, K. Kalantar-Zadeh and T. Daeneke, *Nanoscale Adv.*, 2016, **8**, 16276–16283.
- 151 M. C. Rodríguez González, R. Sasikumar and S. De Feyter, *Surf. Rev. Lett.*, 2021, **28**, 2140002.
- 152 Y. Zhao, S. Ippolito and P. Samorì, *Adv. Opt. Mater.*, 2019, **7**, 1900286.
- 153 G. R. Bhimanapati, Z. Lin, V. Meunier, Y. Jung, J. Cha, S. Das, D. Xiao, Y. Son, M. S. Strano and V. R. Cooper, *ACS Nano*, 2015, **9**, 11509–11539.
- 154 K. Shi, J. Li, Y. Xiao, L. Guo, X. Chu, Y. Zhai, B. Zhang, D. Lu and F. Rosei, *ACS Appl. Mater. Interfaces*, 2020, **12**, 31382–31391.
- 155 R. Nur, T. Tsuchiya, K. Toprasertpong, K. Terabe, S. Takagi and M. Takenaka, *Commun. Mater.*, 2020, **1**, 103.
- 156 F. F. Masouleh and N. Das, Application of Metal-Semiconductor-Metal Photodetector in High-Speed Optical Communication Systems, in *Advances in Optical Communication*, ed. N. Das, 2014, DOI: [10.5772/58997](https://doi.org/10.5772/58997).
- 157 W. Zhi, Q. Quan, P. Yu and Y. Jiang, *Micromachines*, 2020, **11**, 65.
- 158 W. Wu, Q. Zhang, X. Zhou, L. Li, J. Su, F. Wang and T. Zhai, *Nano Energy*, 2018, **51**, 45–53.
- 159 L. Han, M. Yang, P. Wen, W. Gao, N. Huo and J. Li, *Nanoscale Adv.*, 2021, **3**, 2657–2665.
- 160 L. Zeng, L. Tao, C. Tang, B. Zhou, H. Long, Y. Chai, S. P. Lau and Y. H. Tsang, *Sci. Rep.*, 2016, **6**, 20343.
- 161 Q. Qiu and Z. Huang, *Adv. Mater.*, 2021, **33**, 2008126.
- 162 H. K. Sandhu, J. W. John, A. Jakhar, A. Sharma, A. Jain and S. Das, *Adv. Mater. Interfaces*, 2022, **9**, 2102200.
- 163 C. Xie, C. Mak, X. Tao and F. Yan, *Adv. Funct. Mater.*, 2017, **27**, 1603886.
- 164 J. Kublitski, A. Fischer, S. Xing, L. Baisinger, E. Bittrich, D. Spoltore, J. Benduhn, K. Vandewal and K. Leo, *Nat. Commun.*, 2021, **12**, 4259.
- 165 D. Kannichankandy, P. M. Pataniya, C. K. Zankat, M. Tannarana, V. M. Pathak, G. K. Solanki and K. D. Patel, *Appl. Surf. Sci.*, 2020, **524**, 146589.
- 166 Y. Xin, X. Wang, Z. Chen, D. Weller, Y. Wang, L. Shi, X. Ma, C. Ding, W. Li and S. Guo, *ACS Appl. Mater. Interfaces*, 2020, **12**, 15406–15413.
- 167 D. Wu, J. Guo, C. Wang, X. Ren, Y. Chen, P. Lin, L. Zeng, Z. Shi, X. J. Li and C.-X. Shan, *ACS Nano*, 2021, **15**, 10119–10129.
- 168 D. Wu, C. Guo, Z. Wang, X. Ren, Y. Tian, Z. Shi, P. Lin, Y. Tian, Y. Chen and X. Li, *Nanoscale*, 2021, **13**, 13550–13557.
- 169 C. Jia, X. Huang, D. Wu, Y. Tian, J. Guo, Z. Zhao, Z. Shi, Y. Tian, J. Jie and X. Li, *Nanoscale*, 2020, **12**, 4435–4444.
- 170 M. Alamri, M. Gong, B. Cook, R. Goul and J. Z. Wu, *ACS Appl. Mater. Interfaces*, 2019, **11**, 33390–33398.
- 171 C. Zhou, S. Raju, B. Li, M. Chan, Y. Chai and C. Y. Yang, *Adv. Funct. Mater.*, 2018, **28**, 1802954.
- 172 J. Hei, X. Li, S. Wu, P. Lin, Z. Shi, Y. Tian, X. Li, L. Zeng, X. Yu and D. Wu, *ACS Appl. Mater. Interfaces*, 2023, **15**, 12052–12060.
- 173 E.-X. Ding, P. Liu, H. H. Yoon, F. Ahmed, M. Du, A. M. Shafi, N. Mehmood, E. I. Kauppinen, Z. Sun and H. Lipsanen, *ACS Appl. Mater. Interfaces*, 2023, **15**, 4216–4225.
- 174 Z. Dan, B. Yang, Q. Song, J. Chen, H. Li, W. Gao, L. Huang, M. Zhang, M. Yang and Z. Zheng, *ACS Appl. Mater. Interfaces*, 2023, **15**, 18101–18113.
- 175 H. Xu, J. Zhu, G. Zou, W. Liu, X. Li, C. Li, G. H. Ryu, W. Xu, X. Han and Z. Guo, *Nano-Micro Lett.*, 2020, **12**, 1–14.
- 176 M. Patel, P. M. Pataniya and C. Sumesh, *Mater. Res. Bull.*, 2022, **145**, 111518.
- 177 M. Patel, P. M. Pataniya, V. Patel, C. Sumesh and D. J. Late, *Sol. Energy*, 2020, **206**, 974–982.
- 178 S. Kapatel and C. Sumesh, *Opt. Mater.*, 2022, **129**, 112537.
- 179 G. H. Shin, C. Park, K. J. Lee, H. J. Jin and S.-Y. Choi, *Nano Lett.*, 2020, **20**, 5741–5748.
- 180 A. Abdullah Ripain, N. Zulkifli, C. Tan, W. Abd Majid and R. Zakaria, *Opt. Quantum Electron.*, 2022, **54**, 847.
- 181 J. Yuan, S. Zhou, B. Xiao, L. Bao, Z. Ai, Y. Shen, G. Ran and Q. Cheng, *ACS Appl. Nano Mater.*, 2023, **6**, 4594–4601.
- 182 M. Zumuukhorol, Z. Khurelbaatar, D.-H. Kim, K.-H. Shim, V. Janardhanam, V. R. Reddy and C.-J. Choi, *Vacuum*, 2023, **209**, 111746.
- 183 C. Zhou, S. Zhang, Z. Lv, Z. Ma, C. Yu, Z. Feng and M. Chan, *npj 2D Mater. Appl.*, 2020, **4**, 46.
- 184 W. Deng, Y. Chen, C. You, B. An, B. Liu, S. Li, Y. Zhang, H. Yan and L. Sun, *2D Mater.*, 2018, **5**, 045027.
- 185 K.-N. Lee, S. Bang, N. T. Duong, S. J. Yun, D. Y. Park, J. Lee, Y. C. Choi and M. S. Jeong, *ACS Appl. Mater. Interfaces*, 2019, **11**, 20257–20264.
- 186 W. Li, Q. Jia, H. Dong, Z. A. Wang, Y. Wang, Y. Wu, X. Zhao, Z. Chen and S. Wang, *ACS Appl. Nano Mater.*, 2022, **6**, 512–522.



- 187 P. Hu, J. Zhang, M. Yoon, X.-F. Qiao, X. Zhang, W. Feng, P. Tan, W. Zheng, J. Liu and X. Wang, *Nano Res.*, 2014, **7**, 694–703.
- 188 Y. Wen and Z. Wu, *Solid-State Electron.*, 2023, **209**, 108765.
- 189 S. Pak, S. Jang, T. Kim, J. Lim, J. S. Hwang, Y. Cho, H. Chang, A. R. Jang, K. H. Park and J. Hong, *Adv. Mater.*, 2021, **33**, 2102091.
- 190 X. Yu, X. Yu, L. Chen, J. Zhang, Y. Long, L. Zhe, J. Hu, H. Zhang and Y. Wang, *Opt. Mater.*, 2018, **84**, 490–497.
- 191 S. Roy, Aastha, K. A. Deo, K. Dey, A. K. Gaharwar and A. Jaiswal, *ACS Appl. Mater. Interfaces*, 2023, **15**, 35753–35787.
- 192 T. Scharl, G. Binder, X. Chen, T. Yokosawa, A. Cadranel, K. C. Knirsch, E. Spiecker, A. Hirsch and D. M. Guldi, *J. Am. Chem. Soc.*, 2022, **144**, 5834–5840.

