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# Moisture-triggered proton conductivity switching in metal–organic frameworks: role of coordinating solvents†

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Metal–organic frameworks are a good platform for investigating the correlation of structures with physical properties due to the facile coordination environment changes and their responsive structures to external stimuli such as pressure, temperature, and gas sorption. In this study, we report a proton conductivity switching behavior in  $Zn_5FDC$ ,  $[Zn_5(\mu_3-OH)_2(FDC)_4(solvent)_2]$  ( $FDC = 9H$ -fluorene-2,7-dicarboxylate) triggered by relative humidity (RH). Interestingly, depending on the presence and absence of coordinating molecules the MOFs show distinctively different tendencies in their proton conductivity. Two isostructural  $Zn_5FDC$  compounds,  $[Zn_5(\mu_3-OH)_2(DEF)_2(FDC)_4]$  ( $Zn_5FDC-DEF$ ) and  $[Zn_5(\mu_3-OH)_2(FDC)_4]$  ( $Zn_5FDC-OMS$ ;  $OMS =$  open metal site), are prepared, in which the three-dimensional connectivities are identical, but the local structures in the secondary building units (SBUs) are different. In the measurement of humidity-dependent conductivity, both MOFs show a dramatic proton conductivity switching phenomenon (ON/OFF ratio, approximately  $10^8$ ), but the conductivity switching occurs at different RHs for each MOF (above RH 70% in  $Zn_5FDC-DEF$ , and above RH 90% in  $Zn_5FDC-OMS$  at 298 K). During this process water coordination in metal centers leads to their structural transformation into  $Zn_5FDC-H_2O$ , which means that the different coordination structures by the absence/presence of coordination solvents provide different water access environments to metal centers. The computational calculation supports that the structural transformation of  $Zn_5FDC-OMS$  triggered by moisture exposure occurred under higher relative humidity conditions than simple coordination solvent replacement in  $Zn_5FDC-DEF$ . This study proves that the coordination solvents play a role in conductivity variation, and it provides a new design strategy for functional solid-state proton conductors.

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## Introduction

Proton-conductive materials play a pivotal role in proton exchange membrane fuel cells (PEMFC),<sup>1,2</sup> electrolyzers,<sup>3</sup> and

$CO_2$  converters,<sup>4,5</sup> facilitating the generation of clean and efficient electric energy. These applications offer promising renewable energy solutions to mitigate the environmental pollution caused by the combustion of fossil fuels.<sup>6</sup> In addition to well-established proton-conductive materials such as Nafion<sup>7</sup> and oxides,<sup>8</sup> recent attention has been drawn to metal–organic frameworks (MOFs)<sup>9</sup> due to their remarkable design flexibility, high porosity, and crystalline structure. MOFs are emerging as versatile platforms capable of precisely controlling proton conductivity.<sup>10–12</sup> Various methods have been explored to enhance proton conductivity within MOFs, focusing on the induction of continuous hydrogen bonding networks within the MOF pores.<sup>13</sup> These methods involve the decoration of metal centers<sup>14,15</sup> or ligand struts with hydrophilic functional groups,<sup>16,17</sup> as well as the incorporation of suitable guest molecules<sup>18,19</sup> or ions<sup>20</sup> into the MOF pores. These strategies have resulted in the development of MOFs exhibiting exceptional proton conductivity, exceeding  $10^{-3}$  S cm<sup>-1</sup>, under diverse temperature and humidity conditions.<sup>21–26</sup>

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Beyond their role as traditional solid-state proton conductors, MOFs exhibit intriguing proton-conductivity switching behavior in response to external stimuli.<sup>27,28</sup> This makes MOFs more fascinating, with potential applications in sensors,<sup>29,30</sup> memory devices,<sup>31–33</sup> and display technologies.<sup>34</sup> The accumulated knowledge derived from studies on the structural dynamics of MOFs under external stimuli, including electric fields,<sup>31,35</sup> guest molecules,<sup>36</sup> and light,<sup>37–39</sup> holds promise for guiding research towards achieving switchable proton conductivity in MOFs. Over the past decade, research on MOFs that demonstrate proton conductivity switching behavior has been growing. Notably, studies focusing on moisture-induced proton conductivity switching in MOFs have identified water molecules as suitable triggers due to their solid donor–acceptor capabilities and ease of control. Moreover, since practical applications like PEMFCs necessitate structural stability in high humidity environments,<sup>40</sup> the investigation of proton conductivity switching MOFs induced by moisture has evolved from understanding reversible proton transport pathways<sup>41</sup> to employing reversible MOFs capable of enhancing proton conductivity under humid conditions.<sup>27</sup> Among these MOFs, Li and Chen<sup>42</sup> reported a novel self-adaptive MOF, BUT-8(Cr)A, which exhibits proton conductivity switching behavior as water molecules induce structural flexibility. BUT-8(Cr)A, with a high density of sulfonic acid groups on its channel surface, ensures a smooth proton conduction pathway across a wide range of humidity conditions. Due to its structural flexibility, BUT-8(Cr)A demonstrates a self-adaptive proton conductive mechanism. The contracted pores gradually open upon water adsorption, maintaining hydrogen bond networks even at low relative humidity. Notably, a significant shift in proton conductivity, from  $4.19 \times 10^{-6}$  to  $7.61 \times 10^{-2}$  S cm<sup>-1</sup>, was observed as relative humidity increased from 11% to 100% at 25 °C, showing a high ON/OFF ratio of  $10^4$ . Research on such proton-conductivity switching MOFs is still in its early stages, and understanding the dynamics of proton transfer and the role of triggers remains a significant challenge. Additionally, the design and synthesis of MOFs with excellent stimulus responsiveness, including high conductivity, a high ON/OFF ratio, and stable cycling ability, continue to be areas of active investigation.<sup>27</sup>

Herein, we report moisture-triggered switchable proton conductivity in Zn-based MOFs,  $[\text{Zn}_5(\mu_3\text{-OH})_2(\text{FDC})_4]$  ( $\text{Zn}_5\text{FDC}$ , FDC = 9*H*-fluorene-2,7-dicarboxylate), which show a dramatic conductivity change of more than  $10^8$  S cm<sup>-1</sup> within a narrow relative humidity range. Interestingly, depending on the presence and absence of coordinating molecules in secondary building units (SBUs) of  $\text{Zn}_5\text{FDC}$ , the conductivity switching occurs in different ranges of relative humidity (RH). X-ray diffraction studies and theoretical calculation results explain these phenomena: during the proton conductivity measurement,  $\text{Zn}_5\text{FDC}$  MOFs with and without coordinating molecules undergo a structural transformation upon water coordination, and the difference in the coordination environment affects the threshold structure transformation triggered by  $\text{H}_2\text{O}$  molecules. This study provides empirical data and theoretical

substantiation for understanding the influence of coordinating solvents on the proton conduction behaviour in MOFs, contributing to the design strategy for a new functional proton conductor.

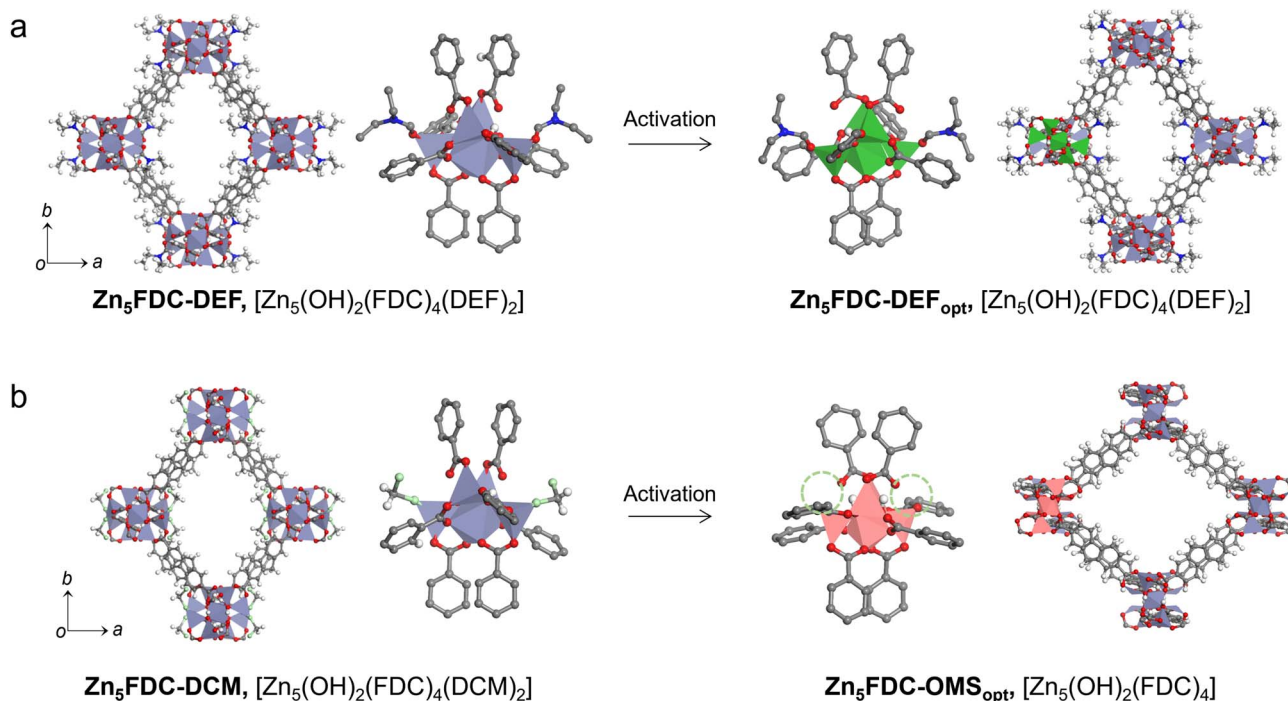
## Results and discussion

### Synthetic strategies of MOFs with/without coordinating solvents

$\text{Zn}_5\text{FDC}$ ,  $[\text{Zn}_5(\mu_3\text{-OH})_2(\text{FDC})_4(\text{solvent})_2]$ , is an appropriate structure for studying the relationship between local structures and properties influenced by different coordinating solvents within the same frameworks (Scheme 1).  $\text{Zn}_5\text{FDC}$  is composed of pentanuclear Zn clusters coordinating with two solvent molecules and four FDC<sup>2-</sup> ligands. The coordinating solvents in  $\text{Zn}_5\text{FDC}$  could be adjusted depending on the synthetic strategy, resulting in  $\text{Zn}_5\text{FDC-DEF}$ ,  $[\text{Zn}_5(\mu_3\text{-OH})_2(\text{FDC})_4(\text{DEF})_2]$ , adopting *N,N*-diethylformamide (DEF) as a strong coordinating solvent, and  $\text{Zn}_5\text{FDC-DCM}$ ,  $[\text{Zn}_5(\mu_3\text{-OH})_2(\text{FDC})_2(\text{DCM})_2]$ , utilizing dichloromethane (DCM) as a weak coordinating solvent.

$\text{Zn}_5\text{FDC-DEF}$  is synthesized *via* a solvothermal reaction using  $\text{H}_2\text{FDC}$  and  $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  dissolved in DEF at 85 °C for 48 h. Single-crystal structure analysis revealed that  $\text{Zn}_5\text{FDC-DEF}$  has the *Pccn* orthorhombic space group, exhibiting pcu net topology composed of  $\text{Zn}_5(\text{OH})_2(\text{DEF})_2$  clusters as octahedral SBUs and FDC<sup>2-</sup> ligands (Table S1, S2, and Fig. S1†). In the pentanuclear Zn clusters, center Zn has octahedral coordination geometry with six oxygen atoms, where two of them come from bridging  $\mu_3\text{-OH}$  and the others are carboxylate oxygen of the ligand. Meanwhile outer four Zn atoms have tetrahedral coordination geometry, where two of the Zn metals have one coordinating DEF solvent, one bridging  $\mu_3\text{-OH}$ , and two carboxylate oxygen of the ligand. In the thermogravimetric analysis,  $\text{Zn}_5\text{FDC-DEF}$  is stable until 400 °C, and the coordinate DEF solvents continuously released around 200–400 °C, exhibiting 6% of weight loss (Fig. S2†). This weight loss corresponding to DEF molecules was also observed after conducting a solvent exchange process with DCM, which was also confirmed through nuclear magnetic resonance (NMR) studies after digesting the MOF samples (Fig. S3†). This means that preparing  $\text{Zn}_5\text{FDC}$  with OMSs is challenging through solvent exchange as well as activation of  $\text{Zn}_5\text{FDC-DEF}$ . Therefore, we tried to introduce a weak coordinating solvent into  $\text{Zn}_5\text{FDC}$  through a solvent-induced structural transformation approach.<sup>43,44</sup> By adjusting the temperature and concentration of metal and ligand based on the synthetic condition of  $\text{Zn}_5\text{FDC-DEF}$ , we obtained the other types of MOFs,  $(\text{H}_3\text{O})_2[\text{Zn}_8(\mu_3\text{-O})_3(\text{FDC})_6]$  ( $\text{Zn}_8\text{FDC}$ ) and  $[\text{Zn}_7(\mu_3\text{-OH})_3(\text{FDC})_6(\text{H}_3\text{O})]$  ( $\text{Zn}_7\text{FDC}$ ), as a mixture. The structures of  $\text{Zn}_8\text{FDC}$  and  $\text{Zn}_7\text{FDC}$  analysed by single crystal X-ray diffraction (SCD) are described in the ESI (Table S4–S6 and Fig. S4, S5)†. Interestingly, both phases converged into  $\text{Zn}_8\text{FDC}$  *via* the reassembly of the metal cluster after immersion in DEF solvent for three days (Fig. S6†). It exhibits a similar trend to previously reported solvent-mediated interpenetration transitions (Fig. S7†).<sup>45</sup>

Subsequently,  $\text{Zn}_8\text{FDC}$  underwent additional solvent-induced structure transformation into  $\text{Zn}_5\text{FDC-DCM}$  during



**Scheme 1** Structural transformation of (a)  $\text{Zn}_5\text{FDC-DEF}$  and (b)  $\text{Zn}_5\text{FDC-DCM}$  upon activation. The activated structures,  $\text{Zn}_5\text{FDC-DEF}_{\text{opt}}$  and  $\text{Zn}_5\text{FDC-OMS}_{\text{opt}}$ , were obtained by optimizing the geometry of  $\text{Zn}_5\text{FDC-DEF}$  and  $\text{Zn}_5\text{FDC-DCM}$  after removing coordinating DCM, respectively. Steel blue: Zn, light green: Cl, grey: C, red: O, blue: N, white: H.

the solvent exchange process with DCM, which is isostructural with  $\text{Zn}_5\text{FDC-DEF}$  (Fig. 1, S8 and Table S1, S3<sup>†</sup>). Both have similar cell parameters except for the  $a$ -axis due to the size difference of the coordinating solvent. To understand the structure transformation mechanism, we compared the simplified networks based on SCD analysis using the oxo molecules of  $\text{Zn}_8\text{FDC}$  and the hydroxy molecules of  $\text{Zn}_5\text{FDC-DCM}$  as dots and the organic linkers as lines (Fig. S9<sup>†</sup>). The structural transformation was induced by the loss of metals and ligands and the transition of oxo groups to hydroxy groups in SBUs in DCM. Through this transformation,  $\text{Zn}_5\text{FDC}$  with weak coordinating solvents, namely  $\text{Zn}_5\text{FDC-DCM}$ , was successfully prepared (Fig. 1b).

### Water sorption and structural transformation

While  $\text{Zn}_5\text{FDC-DEF}$  retains its coordinating DEF molecules during the DCM exchange and activation processes,  $\text{Zn}_5\text{FDC-DCM}$  readily releases its coordinating DCM molecules, resulting in the formation of  $\text{Zn}_5\text{FDC-OMS}$  during the activation process at 120 °C under vacuum for 16 h. In Fig. 2a, Fourier transform infrared (FT-IR) spectra support these phenomena; in  $\text{Zn}_5\text{FDC-OMS}$  the C–Cl stretch at 730  $\text{cm}^{-1}$  disappears, but  $\text{Zn}_5\text{FDC-DEF}_{\text{DCM}}$  ( $\text{Zn}_5\text{FDC-DEF}$  soaked in DCM) and  $\text{Zn}_5\text{FDC-DEF}_{\text{act}}$  ( $\text{Zn}_5\text{FDC-DEF}$  heated at 120 °C under vacuum) still show C=O<sub>DEF</sub> stretch at 1650  $\text{cm}^{-1}$ . To examine the behaviours of  $\text{Zn}_5\text{FDC-DEF}$  and  $\text{Zn}_5\text{FDC-OMS}$  in moisture, the structures of both MOFs were monitored by X-ray powder diffraction (XRPD) after exposure to RH 95% at 25 °C for 48 h. To prevent dew formation and maintain the RH during this experiment, a saturated potassium nitrate solution<sup>46</sup> was used inside

a sealed container. As shown in Fig. 2a, upon exposure to moisture, a coordinating water peak appears around 3600  $\text{cm}^{-1}$  for both MOFs,  $\text{Zn}_5\text{FDC-H}_2\text{O}_{\text{DEF}}$  and  $\text{Zn}_5\text{FDC-H}_2\text{O}_{\text{OMS}}$ , suggesting that water molecules coordinate with the SBUs. Particularly, the replacement of coordinating molecules in  $\text{Zn}_5\text{FDC-DEF}$  is supported by the reduced intensity for the C=O<sub>DEF</sub> stretch at 1650  $\text{cm}^{-1}$ .  $\text{Zn}_5\text{FDC-H}_2\text{O}_{\text{DEF}}$  and  $\text{Zn}_5\text{FDC-H}_2\text{O}_{\text{OMS}}$  exhibit the same XRPD patterns (Fig. 2b), indicating that  $\text{Zn}_5\text{FDC-DEF}$  and  $\text{Zn}_5\text{FDC-OMS}$  converge into the same structure,  $\text{Zn}_5\text{FDC-H}_2\text{O}$ . When compared to the XRPD patterns of the original structures, it can be observed that the coordinating H<sub>2</sub>O molecules enable the formation of hydrogen bonds with surrounding oxygen in the metal cluster, inducing overall structure shrinkage. It is worth noting that even after undergoing the activation and subsequent water adsorption processes, the metal–ligand coordination remains intact, as shown in FT-IR spectra; the peaks for C–O<sub>carboxylate</sub> around 1500–1600  $\text{cm}^{-1}$  exhibit no alteration.

After activation processes SCD analysis for  $\text{Zn}_5\text{FDC-DEF}$  and  $\text{Zn}_5\text{FDC-DCM}$  failed due to their low crystallinity. In order to predict structural changes during the activation, geometry optimization was performed on  $\text{Zn}_5\text{FDC-DEF}$  and  $\text{Zn}_5\text{FDC-DCM}$  removing coordinating DCM using Materials Studio.<sup>47</sup> The simulated XRPD patterns obtained from optimized structures (denoted as ‘opt’ in a subscript after their names such as  $\text{Zn}_5\text{FDC-OMS}_{\text{opt}}$  and  $\text{Zn}_5\text{FDC-DEF}_{\text{opt}}$ ) were compared with empirical XRPD patterns (Fig. S10<sup>†</sup>). The geometry optimization clearly supports that the relatively broader peak of  $\text{Zn}_5\text{FDC-OMS}$  (e.g.  $\text{Zn}_5\text{FDC-DCM}$  after activation) than  $\text{Zn}_5\text{FDC-DEF}$  in XRPD patterns is derived from an anisotropic peak shift, where its

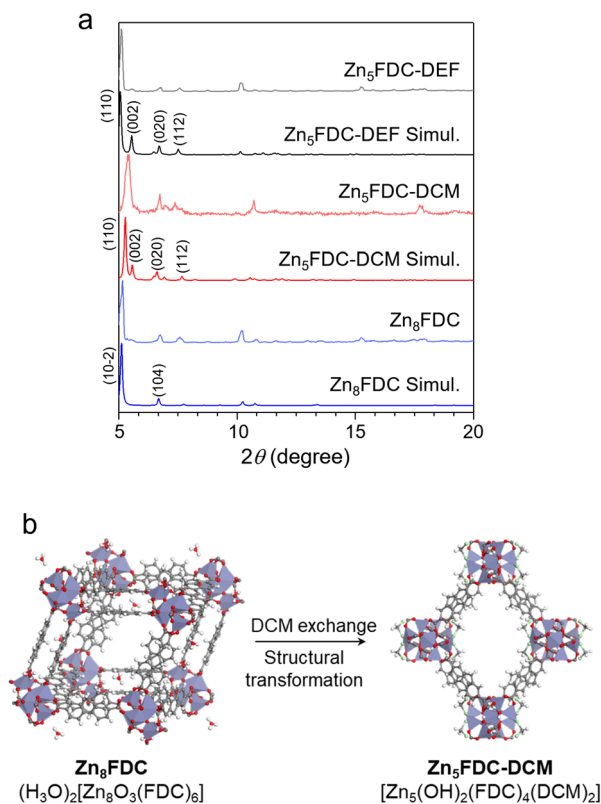


Fig. 1 (a) Comparison of XRPD patterns of Zn<sub>8</sub>FDC, Zn<sub>5</sub>FDC-DCM, and Zn<sub>5</sub>FDC-DEF with their simulated patterns from single crystal structures. (b) Structural transformation of Zn<sub>8</sub>FDC into Zn<sub>5</sub>FDC-DCM during solvent exchange with DCM. Steel blue: Zn, light green: Cl, grey: C, red: O, white: H.

direction in the same plane for Zn<sub>5</sub>FDC-OMS and Zn<sub>5</sub>FDC-DEF is opposite. For instance, (110) and (002) planes for Zn<sub>5</sub>FDC-OMS shift to lower and higher angles than Zn<sub>5</sub>FDC-DCM, respectively, whereas those are opposite in Zn<sub>5</sub>FDC-DEF (Fig. S11†). It implies that O–Zn–O bond angles attaching the coordinating solvents to Zn changed depending on the presence of coordinating solvents during the activation process, as a result, the arrangement of the lattice also changed accordingly (Table S7†). Therefore, the distance of the OMS and the hydrogen of benzene and the oxygen of carboxylate in the neighbouring ligand became closer to stabilize the geometry of the OMS in SBUs of Zn<sub>5</sub>FDC-OMS (Fig. S12†). In contrast, since coordinating DEF molecules in Zn<sub>5</sub>FDC-DEF were bonded to SBUs even after the activation process, only small lattice changes occurred upon bending and rotating of the ligand (Fig. S13†). These local structural differences, with and without coordinating solvent molecules, can lead to a trending preference for H<sub>2</sub>O coordination to SBUs, which will play a key role in proton conductivity switching.

Prior to conducting proton conductivity measurements, water vapor sorption experiments were performed on both Zn<sub>5</sub>FDC-DEF and Zn<sub>5</sub>FDC-OMS. Both MOFs exhibited water uptake at 298 K (Fig. 3), despite being revealed as nonporous materials in N<sub>2</sub> sorption isotherms at 77 K (Fig. S14†). While Zn<sub>5</sub>FDC-OMS displayed a linear increase of water uptake as RH



Fig. 2 (a) FT-IR spectra of Zn<sub>5</sub>FDC-DCM, Zn<sub>5</sub>FDC-OMS, Zn<sub>5</sub>FDC-H<sub>2</sub>O<sub>OMS</sub>, Zn<sub>5</sub>FDC-DEF<sub>DCM</sub>, Zn<sub>5</sub>FDC-DEF<sub>act</sub>, and Zn<sub>5</sub>FDC-H<sub>2</sub>O<sub>DEF</sub>. (b) The XRPD patterns of activated and water-exposed structures for Zn<sub>5</sub>FDC-OMS and Zn<sub>5</sub>FDC-DEF.

increased, Zn<sub>5</sub>FDC-DEF exhibited a stepwise uptake isotherm around RH 90%, which might correspond to structural transformation. Additionally, the difference in total uptake amounts between Zn<sub>5</sub>FDC-DEF (131 cc g<sup>-1</sup>) and Zn<sub>5</sub>FDC-OMS (168 cc g<sup>-1</sup>) can be attributed to the presence or absence of coordinating solvents. The desorption profile having a large hysteresis reveals that the adsorbed water was not completely removed from Zn<sub>5</sub>FDC-OMS and Zn<sub>5</sub>FDC-DEF, implying that water molecules strongly bound to the metal center are difficult to remove. The second cycle of water sorption conducted after evacuation provides indirect evidence for water coordination through a similar reduction in adsorption capacity (50 cc g<sup>-1</sup>, 3 mol<sup>-1</sup>), which is approximately consistent with the remaining water in the first cycle.

### Switchable proton conductivity

We examined proton conductivity for Zn<sub>5</sub>FDC-DEF and Zn<sub>5</sub>FDC-OMS and clarified the effects of moisture adsorption tendencies and coordinating solvents on their proton conductivity. To

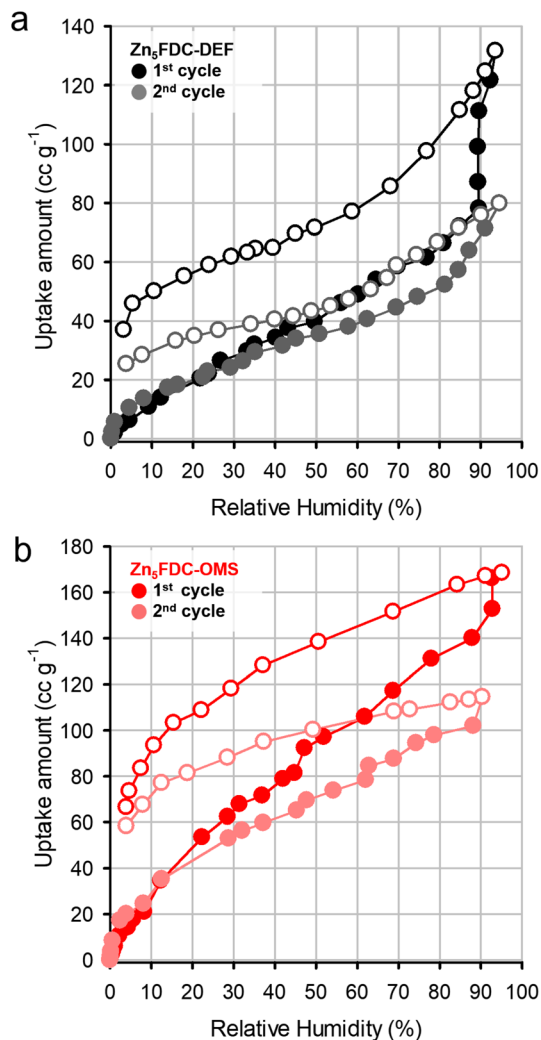


Fig. 3 Water vapor sorption isotherms of (a)  $\text{Zn}_5\text{FDC-DEF}$  and (b)  $\text{Zn}_5\text{FDC-OMS}$  at 298 K.

verify the proton conductivity, the impedance measurements for both MOFs were conducted at different RHs ranging from 30% to 95% and temperatures (Fig. 4a, S15, and Table S9†). At RH 30% and 298 K, the proton conductivities of  $\text{Zn}_5\text{FDC-OMS}$

and  $\text{Zn}_5\text{FDC-DEF}$  are negligible ( $<10^{-12} \text{ S cm}^{-1}$ ). Although humidity increased from 30 to 90%, the conductivity of  $\text{Zn}_5\text{FDC-OMS}$  exhibited only a modest one-order increase, indicating the conductivity values of  $3.33 \times 10^{-12} \text{ S cm}^{-1}$  at 298 K and RH 90%. In contrast,  $\text{Zn}_5\text{FDC-DEF}$  exhibited a continuous improvement of conductivity, and a steep jump occurred above RH 70%, reaching  $8.97 \times 10^{-6} \text{ S cm}^{-1}$  at RH 90%. This value improved more to  $2.49 \times 10^{-4} \text{ S cm}^{-1}$  at RH 95%.  $\text{Zn}_5\text{FDC-DEF}$  has the difference between the proton conductivity switching point and steep jump at water sorption because it could exhibit self-adaptive conductivity based on the interaction with water molecules even before its structure is completely transformed by moisture (Fig. S16†). Notably,  $\text{Zn}_5\text{FDC-OMS}$  also exhibits a comparable switching behaviour around RH 90%, showing different conductivity switching points from  $\text{Zn}_5\text{FDC-DEF}$ . Proton conductivity switching can be attributed to the induced structural transformation by pore-filling  $\text{H}_2\text{O}$  as well as coordinating  $\text{H}_2\text{O}$ , which is mainly responsible for proton conductivity due to its acidic properties. Thus, different conductivity switching points in  $\text{Zn}_5\text{FDC-DEF}$  and  $\text{Zn}_5\text{FDC-OMS}$  can be expected to be different interaction energy of  $\text{H}_2\text{O}$  with MOFs at different RH.

In order to understand these phenomena, the interaction energy between water and water as well as water and MOFs at RH 30, 70, and 95% was calculated through RASPA<sup>48</sup> of NVT simulations (Table S8†). This strategy is based on the assumption that proton conductive  $\text{Zn}_5\text{FDC-H}_2\text{O}$  would have a strong interaction with moisture from a certain relative pressure after structural transformation compared to the starting materials, and this is known through a comparison of interaction energy according to relative pressure. At RH 30%,  $\text{Zn}_5\text{FDC-OMS}$  exhibits relatively more favourable water–water interaction energy ( $-17.05 \text{ kJ mol}^{-1}$ ) and relatively more unfavourable water–MOF interaction energy ( $-15.36 \text{ kJ mol}^{-1}$ ) than  $\text{Zn}_5\text{FDC-DEF}$  ( $-7.319 \text{ kJ mol}^{-1}$  and  $-22.49 \text{ kJ mol}^{-1}$ , respectively). It implies that  $\text{Zn}_5\text{FDC-OMS}$  is relatively stable against structure transformation to  $\text{Zn}_5\text{FDC-H}_2\text{O}$  induced by moisture exposure compared to  $\text{Zn}_5\text{FDC-DEF}$  at low relative humidity. This tendency is not reversed, even though the energy gap between MOFs is decreased at 70% RH. Thus, the proton conductivity switching behaviour of  $\text{Zn}_5\text{FDC-DEF}$  could occur under lower



Fig. 4 (a) Proton conductivity of  $\text{Zn}_5\text{FDC-DEF}$  (black) and  $\text{Zn}_5\text{FDC-OMS}$  (red) at 298 K under variable humidity. (b) The activation energy ( $E_a$ ) from the temperature-dependent conductivity of  $\text{Zn}_5\text{FDC-DEF}$  (black) and  $\text{Zn}_5\text{FDC-OMS}$  (red and orange) measured at RH 95%. (c) Cycling test  $\text{Zn}_5\text{FDC-OMS}$  between 30 and 95% RH at 298 K.

pressure conditions compared to Zn<sub>5</sub>FDC-OMS, and DEF to water replacement in Zn<sub>5</sub>FDC-DEF could occur under lower pressure conditions than coordination of water in the OMS of Zn<sub>5</sub>FDC-OMS. The water–water interaction energy was reversed at 95% RH (water–water interaction energy: Zn<sub>5</sub>FDC-OMS;  $-25.63 \text{ kJ mol}^{-1}$ , Zn<sub>5</sub>FDC-DEF;  $-27.13 \text{ kJ mol}^{-1}$ ), which implies that Zn<sub>5</sub>FDC-H<sub>2</sub>O<sub>DEF</sub> has relatively stronger water–water interaction to get a higher conductivity than Zn<sub>5</sub>FDC-H<sub>2</sub>O<sub>OMS</sub>.

Subsequently, the temperature-dependent proton conductivity was measured at RH 95%. As illustrated in Fig. 4b, the proton conductivity of both compounds indicates Arrhenius behaviour. The highest proton conductivity value was  $8.41 \times 10^{-5} \text{ S cm}^{-1}$  for Zn<sub>5</sub>FDC-OMS at 80 °C and RH 95% and  $4.26 \times 10^{-4} \text{ S cm}^{-1}$  for Zn<sub>5</sub>FDC-DEF at 60 °C and RH 95% (Table S9†). Further, increased temperature led to a decrease in the conductivity due to the release of guest water molecules. The activation energy ( $E_a$ ) evaluated from temperature-dependent conductivity for Zn<sub>5</sub>FDC-DEF is 0.11 eV, indicating the Grotthuss mechanism of proton conduction where the proton diffuses from one site to another through the hydrogen bond networks. To estimate the activation energy of Zn<sub>5</sub>FDC-OMS, the temperature-dependent conductivity plot clearly needs to be divided into two segments. In the range from 298 K to 303 K, the Arrhenius fit of the data results in a relatively high  $E_a$  of 2.52 eV, likely associated with the occurrence of a structural transformation.<sup>49</sup> However, the segment from 303 K to 353 K corresponds to the Grotthuss mechanism, featuring a lower  $E_a$  of 0.39 eV. This variation in the water-mediated conductivity in the two systems can be ascribed to the potential exchange of coordinated DEF molecules within Zn<sub>5</sub>FDC-DEF with guest water occurring even at lower RH at 298 K, resulting in the formation of Zn<sub>5</sub>FDC-H<sub>2</sub>O<sub>DEF</sub> and the creation of stable and extensive proton-conducting pathways. Conversely, this exchange is energetically unfavourable in the case of Zn<sub>5</sub>FDC-OMS at low RH, as it already possesses a stable geometric configuration. Therefore, it requires relatively higher temperatures where the water molecules are used not only as carriers to transport protons but also to transform the material. The XRPD patterns before and after the water-mediated proton conductivity studies strongly support a structural conversion (Fig. S17†). Finally, both compounds, Zn<sub>5</sub>FDC-OMS and Zn<sub>5</sub>FDC-DEF, attain the overall same structure (Zn<sub>5</sub>FDC-H<sub>2</sub>O<sub>OMS</sub> or Zn<sub>5</sub>FDC-H<sub>2</sub>O<sub>DEF</sub>).

In the relative humidity cycling measurement at RH 30% and 95%, these transformed materials exhibit stable conductivity over three cycles, indicating structural stability (Fig. 4c and S17†). Furthermore, Zn<sub>5</sub>FDCs demonstrate a superior high on-off ratio of approximately  $10^8$ , compared to other previously reported switchable proton conductive MOFs (Table S10†).

## Conclusions

In this study, we reported two isostructural MOFs, Zn<sub>5</sub>FDC-DEF and Zn<sub>5</sub>FDC-OMS, having different coordinating environments by the presence and absence of coordinating solvents. Both MOFs showed structural transformation upon water uptake under various conditions of relative humidity. The results

showed that both MOFs exhibited different response tendencies depending on the relative humidity and temperature, indicating the effects of the presence of coordinating solvents. Specifically, Zn<sub>5</sub>FDC-DEF demonstrated a dramatic improvement in conductivity above 70% relative humidity. Meanwhile, Zn<sub>5</sub>FDC-OMS showed conductivity at relatively higher humidity levels, RH 90% and maintained its conductivity performance at higher temperatures than Zn<sub>5</sub>FDC-DEF.

The difference in the conductive switching points and proton conductivity depending on the temperature of the two MOFs is interpreted to be due to the structural variations of the starting material, influenced by coordinating solvents. Based on temperature-dependent conductivity measurements, both MOFs exhibited Arrhenius behaviours. Notably, Zn<sub>5</sub>FDC-DEF has relatively lower activation energy, indicating a Grotthuss mechanism where protons diffuse from one location to another through a hydrogen-bonding network. Finally, cycling experiments verified the structural robustness of these MOFs. Our findings suggest that coordinating solvents play a role in modulating proton conductivity within structurally similar MOFs. This work sets the stage for future investigations into how coordinating solvents impact the proton-conductive properties of MOFs.

## Author contributions

H. K. L., Y. O., and J. L. contributed equally to this work. H. K. L. conducted synthesis, characterization, and data analysis. Y. O. conducted water sorption and proton conductivity measurements. J. L. and M. K. performed ligand design and synthesis. S. H. carried out theoretical calculations and analysis. D. M. conducted crystallographic analysis. H. K. L., D.-W. L., and H. R. M. planned the research. All authors contributed to discussions and manuscript writing.

## Conflicts of interest

There are no conflicts to declare.

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## References

- 1 A. Kirubakaran, S. Jain and R. K. Nema, *Renewable Sustainable Energy Rev.*, 2009, **13**, 2430–2440.
- 2 S. M. Haile, *Acta Mater.*, 2003, **51**, 5981–6000.

- 3 M. Ni, M. K. H. Leung and D. Y. C. Leung, *Int. J. Hydrogen Energy*, 2008, **33**, 2337–2354.
- 4 R. J. Lim, M. Xie, M. A. Sk, J.-M. Lee, A. Fisher, X. Wang and K. H. Lim, *Catal. Today*, 2014, **233**, 169–180.
- 5 S. Uhm and Y. D. Kim, *Curr. Appl. Phys.*, 2014, **14**, 672–679.
- 6 *Fuel Cell Technologies Office Multi-Year Research, Development, and Demonstration Plan*, U.S. Department of Energy, 2016.
- 7 C. H. Park, C. H. Lee, M. D. Guiver and Y. M. Lee, *Prog. Polym. Sci.*, 2011, **36**, 1443–1498.
- 8 K. D. Kreuer, *Annu. Rev. Mater. Res.*, 2003, **33**, 333–359.
- 9 H. C. Zhou, J. R. Long and O. M. Yaghi, *Chem. Rev.*, 2012, **112**, 673–674.
- 10 D.-W. Lim and H. Kitagawa, *Chem. Rev.*, 2020, **120**, 8416–8467.
- 11 T. Hou, W. Xu, X. Pei, L. Jiang, O. M. Yaghi and K. A. Persson, *J. Am. Chem. Soc.*, 2022, **144**, 13446–13450.
- 12 M. Sadakiyo and H. Kitagawa, *Dalton Trans.*, 2021, **50**, 5385–5397.
- 13 D.-W. Lim and H. Kitagawa, *Chem. Soc. Rev.*, 2021, **50**, 6349–6368.
- 14 N. C. Jeong, B. Samanta, C. Y. Lee, O. K. Farha and J. T. Hupp, *J. Am. Chem. Soc.*, 2012, **134**, 51–54.
- 15 S.-S. Liu, Z. Han, J.-S. Yang, S.-Z. Huang, X.-Y. Dong and S.-Q. Zang, *Inorg. Chem.*, 2020, **59**, 396–402.
- 16 S. M. Elahi, S. Chand, W.-H. Deng, A. Pal and M. C. Das, *Angew. Chem., Int. Ed.*, 2018, **57**, 6662–6666.
- 17 D. Lee, S. Lee, Y. Son, J. Y. Kim, S. Cha, D. Kwak, J. Lee, J. Kwak, M. Yoon and M. Kim, *Bull. Korean Chem. Soc.*, 2022, **43**, 912–917.
- 18 W. J. Phang, W. R. Lee, K. Yoo, D. W. Ryu, B. Kim and C. S. Hong, *Angew. Chem., Int. Ed.*, 2014, **53**, 8383–8387.
- 19 S. Hwang, E. J. Lee, D. Song and N. C. Jeong, *ACS Appl. Mater. Interfaces*, 2018, **10**, 35354–35360.
- 20 M. Sadakiyo, H. Okawa, A. Shigematsu, M. Ohba, T. Yamada and H. Kitagawa, *J. Am. Chem. Soc.*, 2012, **134**, 5472–5475.
- 21 A. Sharma, J. Lim, S. Jeong, S. Won, J. Seong, S. Lee, Y. S. Kim, S. B. Baek and M. S. Lah, *Angew. Chem., Int. Ed.*, 2021, **60**, 14334–14338.
- 22 S. C. Pal and M. C. Das, *Adv. Funct. Mater.*, 2021, **31**, 2101584.
- 23 D. Gui, W. Duan, J. Shu, F. Zhai, N. Wang, X. Wang, J. Xie, H. Li, L. Chen, J. Diwu, Z. Chai and S. Wang, *CCS Chem.*, 2019, **1**, 197–206.
- 24 Q. Lin, Y. Ye, L. Liu, Z. Yao, Z. Li, L. Wang, C. Liu, Z. Zhang and S. Xiang, *Nano Res.*, 2021, **14**, 387–391.
- 25 G. K. H. Shimizu, J. M. Taylor and S. Kim, *Science*, 2013, **341**, 354–355.
- 26 S. Chand, S. M. Elahi, A. Pal and M. C. Das, *Chem. – Eur. J.*, 2019, **25**, 6259–6269.
- 27 F. Xiang, S. Chen, Z. Yuan, L. Li, Z. Fan, Z. Yao, C. Liu, S. Xiang and Z. Zhang, *JACS Au*, 2022, **2**, 1043–1053.
- 28 C. Li, K. Wang, J. Li and Q. Zhang, *ACS Mater. Lett.*, 2020, **2**, 779–797.
- 29 S.-S. Bao, N.-Z. Li, J. M. Taylor, Y. Shen, H. Kitagawa and L.-M. Zheng, *Chem. Mater.*, 2015, **27**, 8116–8125.
- 30 R. Liu, Y. Liu, S. Yu, C. Yang, Z. Li and G. Li, *ACS Appl. Mater. Interfaces*, 2019, **11**, 1713–1722.
- 31 Z. Yao, L. Pan, L. Liu, J. Zhang, Q. Lin, Y. Ye, Z. Zhang, S. Xiang and B. Chen, *Sci. Adv.*, 2019, **5**, eaaw4515.
- 32 Y. Hiruma, K. Yoshikawa and M.-a. Haga, *Faraday Discuss.*, 2019, **213**, 99–113.
- 33 M.-K. Song, S. D. Namgung, D. Choi, H. Kim, H. Seo, M. Ju, Y. H. Lee, T. Sung, Y.-S. Lee, K. T. Nam and J.-Y. Kwon, *Nat. Commun.*, 2020, **11**, 5896.
- 34 Y. Wang, R. Shen, S. Wang, Q. Chen, C. Gu, W. Zhang, G. Yang, Q. Chen, Y.-M. Zhang and S. X.-A. Zhang, *Chem*, 2021, **7**, 1308–1320.
- 35 L. Pan, G. Liu, H. Li, S. Meng, L. Han, J. Shang, B. Chen, A. E. Platero-Prats, W. Lu, X. Zou and R.-W. Li, *J. Am. Chem. Soc.*, 2014, **136**, 17477–17483.
- 36 R.-L. Liu, W.-T. Qu, B.-H. Dou, Z.-F. Li and G. Li, *Chem.–Asian J.*, 2020, **15**, 182–190.
- 37 K. Müller, J. Helfferich, F. Zhao, R. Verma, A. B. Kanj, V. Meded, D. Bléger, W. Wenzel and L. Heinke, *Adv. Mater.*, 2018, **30**, 1706551.
- 38 H.-Q. Liang, Y. Guo, Y. Shi, X. Peng, B. Liang and B. Chen, *Angew. Chem., Int. Ed.*, 2020, **59**, 7732–7737.
- 39 D.-D. Xu, W.-W. Dong, M.-K. Li, H.-M. Han, J. Zhao, D.-S. Li and Q. Zhang, *Inorg. Chem.*, 2022, **61**, 21107–21114.
- 40 J. Zhao, Z. Tu and S. H. Chan, *Energy*, 2022, **239**, 122270.
- 41 M. Sadakiyo, T. Yamada, K. Honda, H. Matsui and H. Kitagawa, *J. Am. Chem. Soc.*, 2014, **136**, 7701–7707.
- 42 F. Yang, G. Xu, Y. Dou, B. Wang, H. Zhang, H. Wu, W. Zhou, J.-R. Li and B. Chen, *Nat. Energy*, 2017, **2**, 877–883.
- 43 G. K. Kole and J. J. Vittal, *Chem. Soc. Rev.*, 2013, **42**, 1755–1775.
- 44 J.-P. Zhang, P.-Q. Liao, H.-L. Zhou, R.-B. Lin and X.-M. Chen, *Chem. Soc. Rev.*, 2014, **43**, 5789–5814.
- 45 S. B. Choi, H. Furukawa, H. J. Nam, D.-Y. Jung, Y. H. Jhon, A. Walton, D. Book, M. O’Keeffe, O. M. Yaghi and J. Kim, *Angew. Chem., Int. Ed.*, 2012, **51**, 8791–8795.
- 46 L. Greenspan, *J. Res. NBS. A. Phys. Ch.*, 1977, **81**, 89.
- 47 BIOVIA. Dassault Systèmes, *Biovia Materials Studio 2017 R2*, Dassault Systèmes, San Diego, 2017.
- 48 D. Dubbeldam, S. Calero, D. E. Ellis and R. Q. Snurr, *Mol. Simul.*, 2016, **42**, 81–101.
- 49 R. M. P. Colodrero, P. Olivera-Pastor, E. R. Losilla, D. Hernández-Alonso, M. A. G. Aranda, L. Leon-Reina, J. Rius, K. D. Demadis, B. Moreau, D. Villemin, M. Palomino, F. Rey and A. Cabeza, *Inorg. Chem.*, 2012, **51**, 7689–7698.