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CdS nanospheres deposited with NiS₂ was prepared by one-step deep eutectic solvothermal method. The prepared NiS₂/CdS exhibits efficiently visible-light-driven dehydrogenation of biomass intermediate 5-hydroxymethylfurfural (HMF) to one of the most important platform molecules in the organic chemicals industry 2,5-diformylfuran (DFF). The entire photocatalytic process could be achieved in water at ambient condition without using sacrificial agents or organic solvents. The photocatalytic dehydrogenation strategy could be used in other biomass-derivative transformation processes.

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Deep eutectic solvothermal NiS₂/CdS synthesis for the visible-light-driven valorization of the biomass intermediate 5-hydroxymethylfurfural (HMF) integrated with H₂ production†

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Visible-light-driven transformation of biomass to high-valued chemicals is attractive for achieving a low-carbon environment. The solar-catalyzed oxidation of 5-hydroxymethylfurfural (HMF) to 2,5-diformylfuran (DFF) has great industrial potential because DFF is a pivotal precursor in industries. Here, we present a one-step deep eutectic solvothermal technique for the production of NiS₂/CdS nanospheres. Because NiS₂ can act as a cocatalyst enhancing the transfer and separation of photogenerated carriers, the prepared NiS₂/CdS exhibits efficient visible-light-driven dehydrogenation of HMF to H₂ and DFF. DFF and H₂ can be produced at rates of up to 519.64 and 476.60 μmol g⁻¹ h⁻¹, respectively, which are about 10.03 and 9.81 times higher than those of bare CdS under visible light irradiation. Moreover, the photocatalytic process was carried out at ambient temperature and pressure in water without using a sacrificial agent. Both h⁺ and e⁻ are involved in the process of photocatalytic reaction, and the valence band position of the NiS₂/CdS photocatalyst is essential for the high DFF selectivity (95.2%). This work paves a sustainable way to realize the photocatalytic valorization of biomass.

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Introduction

5-Hydroxymethylfurfural (HMF) is one of the most promising chemical platforms derived from cellulosic biomass.^{1–3} It has been widely used as a versatile precursor for the creation of high-value fine chemicals.^{4–7} Among the many derivatives of HMF,^{8–10} 2,5-diformylfuran (DFF) used as a monomer plays an important role in medicine intermediates, organic conductors, and functional polymers.^{11,12} HMF could be oxidized by a thermocatalytic process at high temperature and high pressure.^{13,14} However, the thermocatalytic reaction is energy-intensive and often leads to safety and environmental issues.¹⁵

In recent years, the selective oxidation of HMF to DFF through photocatalytic technology has drawn much attention.^{12,16–18} Yurdakal *et al.* first reported the application of anatase TiO₂ for the catalytic oxidation of HMF in water

under an ultraviolet light source.¹⁹ The majority of the organic products were DFF with a selectivity of just 10% to 25%. Wu *et al.*²⁰ reported the preparation of DFF by the catalytic oxidation of HMF with Nb₂O₅ under visible light using trifluorobenzene as the solvent, and the yield of DFF was 20%. Giannakoudakis *et al.*⁹ used MnO₂ nanorods as a catalyst and acetonitrile as a reaction medium and achieved UV-promoted oxidation of HMF to DFF. HMF was nearly completely converted to DFF without the addition of oxidants and bases. In addition to the aforementioned metal oxides, metal sulfides can also be used as photocatalysts.^{21,22} Among a variety of metal sulfides, cadmium sulfide (CdS) has garnered significant interest due to its appropriate band gap (2.4 eV) for light harvesting and appropriate conduction band edge for various catalytic reactions.^{23,24} However, the photocatalytic activity of pure CdS is insufficient due to the significant recombination of carriers and intense photocorrosion.^{25,26} Cocatalysts can help move electrons and holes to the surface active sites and stop charge carriers from getting back together. They could improve the speed of the reaction and the efficiency of the photocatalytic system.^{27,28} Wang *et al.* reported Ni_xS_y for the electrochemical oxidation of water and found that the adsorption energy of water on Ni_xS_y was close to that of water on IrO₂, an excellent catalyst for water oxidation, even in semiconductor photocatalysis.²⁹ To this end, it is highly possible

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that Ni_xS_y also plays an important role in the photocatalytic process. However, the photocatalytic oxidation of HMF to DFF by CdS using NiS_2 as a cocatalyst is yet to be explored.

Nanostructure synthesis by deep eutectic solvothermal procedure has the advantage of being easy to proceed and the structure and composition can be tuned by adjusting the components of solvent, one-step preparation of the sulfide avoiding the additional oxidation process, and reactants are evenly distributed at the molecular level, thus the obtained nanostructures are homogenous. Therefore, more and more works on deep eutectic solvothermal have been carried out.^{30,31}

In this work, CdS nanospheres deposited with NiS_2 were prepared by a one-step solvothermal method. Highly selective conversion of HMF to DFF was achieved by optimizing the synthesis conditions. *In situ* electron paramagnetic resonance (EPR) trapping experiments revealed that $\cdot C_6H_4O_3$ is the major intermediate radical in the oxidation of HMF. The whole photocatalytic process occurred in water at ambient conditions without using sacrificial agents or organic solvents.

Experimental sections

Chemicals

Polyethylene glycol 200 (PEG 200, AR), thiourea (CH_4N_2S , 99.0%), ethyl alcohol (C_2H_5OH , AR), HMF ($C_6H_6O_3$, 99.0%), DFF ($C_6H_4O_3$, 99.0%), furfural alcohol ($C_5H_6O_2$, 99.0%) and furfural ($C_5H_4O_2$, 99.0%) were purchased from Sinopharm Chemical Reagent Co., Ltd. Cadmium acetate dihydrate ($Cd(Ac)_2 \cdot 2H_2O$, AR) and nickel nitrate hexahydrate ($Ni(NO_3)_2 \cdot 6H_2O$, AR) were bought from J&K Ltd.

Preparation of photocatalysts

Synthesis procedure for nanostructures: PEG-based deep eutectic solvent (DES) as a solvent and sulfur source was obtained based on the previous work.³² Here, the synthesis procedure of CdS is taken as an example. At first, DES was prepared in a 2:1 molar ratio of PEG 200 and thiourea at 80 °C. Then, $Cd(Ac)_2 \cdot 2H_2O$ (0.01 mol) was dissolved in 40 ml of DES and stirred at 60 °C until a homogeneous solution was formed. Next, the mixture was placed into a 50 ml Teflon-lined stainless steel autoclave and kept in an oven at 120 °C for 12 h before cooling in air. The sample was obtained by centrifugal washing with deionized (DI) water and ethanol six times. Lastly, the yellow precipitate was dried in a vacuum at 60 °C for 12 h. NiS_2/CdS composite was prepared using the same procedure as shown in Fig. 1. The detailed procedures for the synthesis of NiS_2/CdS , NiS_2 , and $NiS_2 + CdS$ are given in the ESI section.†

Characterization of photocatalysts

For the characterization of the obtained samples, the following techniques were used. X-ray powder diffraction (XRD) performed on a D/max-2550/PC diffractometer (Rigaku), a scanning electron microscope (SEM), energy dispersive X-ray spectroscopy (EDS), and elemental mapping images on a Hitachi



Fig. 1 A schematic illustration of the formation of NiS_2/CdS photocatalyst.

SU 8010 field emission SEM combined with EDS, X-ray photoelectron spectroscopy (XPS) on an ESCALAB MK X-ray photoelectron spectrometer, morphological and microstructural features using high-resolution transmission electron microscopy (HRTEM JEOL-2010), diffuse reflectance spectroscopy using a UV-vis spectrophotometer (Hitachi U-4100), specific surface areas using the nitrogen adsorption-desorption instrument (JW-BK122 W, China), steady-state photoluminescence (PL) spectra and PL decay on an Edinburgh FLS 980 transient steady-state spectrometer, and the carbon-centered radicals by electron paramagnetic resonance (EPR, Bruker EMXnano, Bruker, Germany). Detailed procedures for the above measurements are provided in the ESI section.†

Catalytic activity and mechanism

Experiments on photocatalytic hydrogen evolution were carried out at room temperature in hermetically sealed Pyrex glass. The produced photocatalyst powder was suspended in 50 mL of HMF aqueous solution in a typical reaction (10 mM). Before irradiation with visible light, the suspension was thoroughly degassed with N_2 for 30 min to completely remove dissolved oxygen. The visible light source was a 300 W Xe lamp (LabSolar II, Perfect Light Co., Ltd) equipped with a blocking filter ($\lambda \geq 420$ nm). Gaseous H_2 was regularly detected by a gas chromatograph (GC-2014C, Shimadzu, Japan), assembled with a TCD detector and ultrapure nitrogen as a carrier gas. The reaction solution was analyzed by HPLC, installed with a 4.6 mm \times 250 mm Shim-pack GWS 5 μ m C18 column, and the wavelength of the UV detector was set at 265 nm. In each measurement, 20 μ L of the solution was taken out, diluted to 2 mL with an eluent, and then filtered through a 0.22 μ m PTFE membrane filter. Two eluting solvents were methanol (A) and 5 mM ammonium formate aqueous solution (B), respectively. The ratio of A to B was 3:7 with a flow rate of 0.6 mL min^{-1} and each separation lasted for 10 min. The quantification was performed from a standard curve of known compound concentrations (Fig. S1†). The DFF yield and DFF selectivity were calculated using the following equations, respectively.

$$DFF \text{ yield}(\%) = \frac{n_{DFF \text{ formation}}}{n_{HMF \text{ initial value}}} \times 100\% \quad (1)$$

$$DFF \text{ selectivity}(\%) = \frac{n_{DFF \text{ formation}}}{n_{HMF \text{ consumption}}} \times 100\% \quad (2)$$

where n is the mole of reactant calculated from the concentration measured by HPLC.

The electrochemical experiments analysis was performed using an electrochemical workstation (Shanghai ChenHua Instrument Company). Electron paramagnetic resonance measurement (EPR) spectrometer (Bruker ELEXSYS E500). Density functional theory (DFT) calculations were carried out using the Vienna *ab initio* simulation (VASP5.4.4) code.³³ Detailed information on the above investigation is given in the ESI section.†

Results and discussion

Catalyst characterization

As shown in Fig. 2a, XRD patterns were used to determine the crystallization and phase purity of pure CdS NPs and NiS₂/CdS composites with different amounts of NiS₂. The major sharp peaks centered at 24.8°, 26.5°, 28.2°, 43.7°, 47.8°, 51.8° and

66.8°, correspond to the classical hexagonal CdS phase (JCPDS no. 41-1049) and show high crystallinity.³⁴ The successful synthesis of NiS₂/CdS is shown by the presence of the labeled emblematic peaks of nickel sulfide (NiS₂, JCPDS, no. 11-0099) in the XRD patterns after NiS₂ was added.^{32,35}

The morphologies and internal microstructure of the catalysts were characterized by SEM and HRTEM. The well-scattered CdS spheres have a rough surface and an average diameter of around 100 nm, as seen in Fig. 2c. Pure CdS and 10% NiS₂/CdS had similar morphologies, with no discernible differences (as shown in Fig. S2†). While pure NiS₂ contained peony-like particles, its subunit sheets consisted of many ultra-thin films (Fig. 2d). Further careful observation of the HRTEM image (Fig. 2e) showed that the decoration of NiS₂ was successful, indicating that the interface between NiS₂ and CdS was tightly connected. The interplanar spacing of 0.33 nm corresponds to the (002) planes of CdS,³⁶ and the compact NiS₂ film with an interplanar spacing of 0.23 nm can be assigned to the (211) facet of NiS₂.³⁷ Elemental mapping pat-

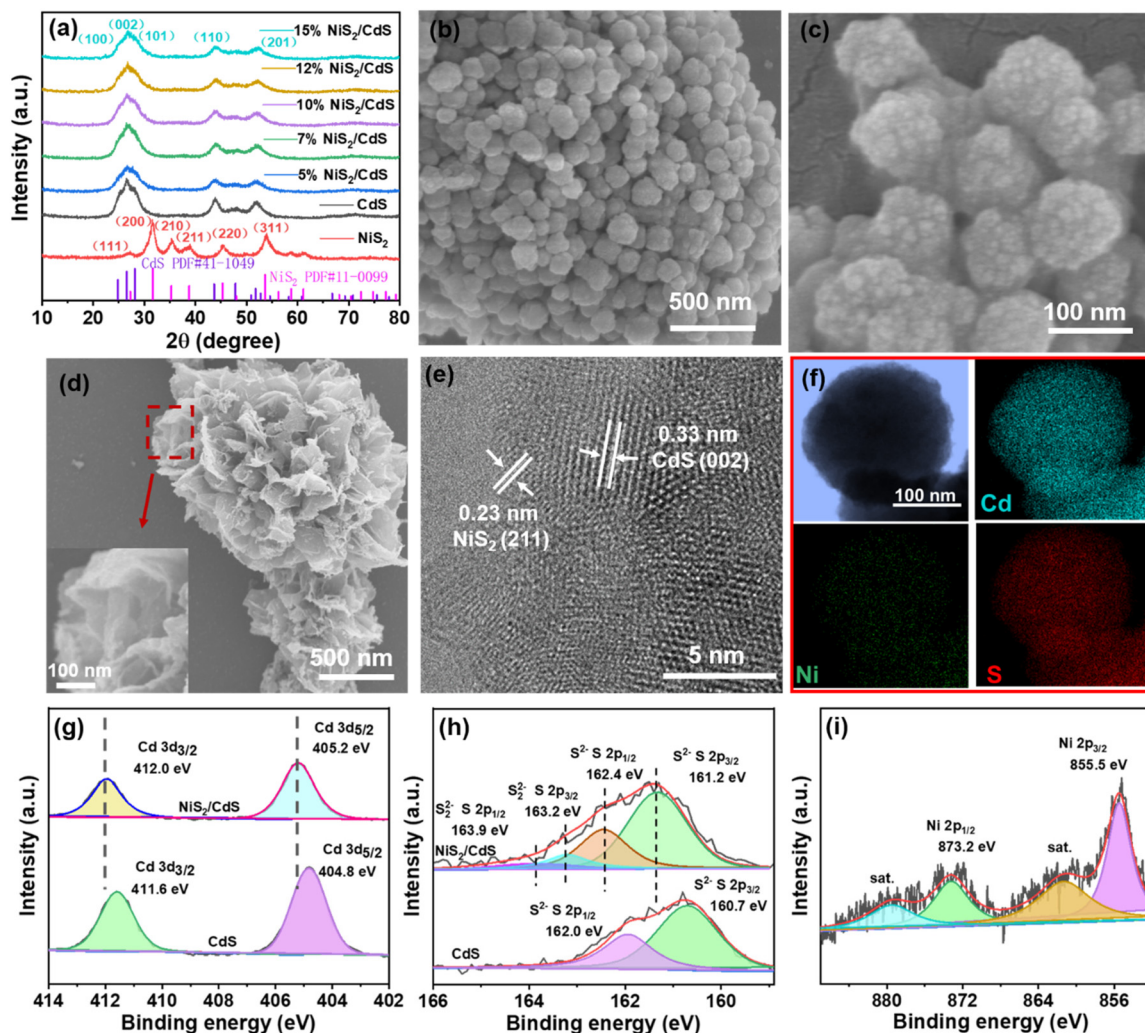


Fig. 2 (a) XRD patterns of the as-prepared CdS, $x\%$ NiS₂/CdS, and NiS₂; (b and c) SEM, (e) HRTEM, (f) element mapping images of the as-prepared 10% NiS₂/CdS; (d) SEM of NiS₂; (g–i) XPS spectra for CdS and 10% NiS₂/CdS.

terns (Fig. 2f) clearly confirmed that Cd, Ni and S elements are uniformly distributed on the nanosphere arrays. The EDS spectrum of the 10% NiS₂/CdS sample is shown in Fig. S3,† further revealing that the sample had clear signals of Cd, Ni, and S elements.

X-ray photoelectron spectroscopy (XPS) study was used to learn more about the surface compositions and electronic states of NiS₂/CdS. The XPS measurement spectra shown in Fig. S4† indicate the presence of Cd, Ni, and S in the corresponding samples, respectively. The characteristic high-resolution XPS spectra of Cd 3d (Fig. 2g) of pure CdS and 10% NiS₂/CdS before the reaction exhibits two binding energy peaks at 411.6 eV, 404.8 eV, and 412.0 eV, 405.2 eV, respectively. In the S 2p spectrum of 10% NiS₂/CdS, the large peak starting from 159 to 165 eV can be deconvoluted into four peaks (Fig. 2h). Those with energies of 163.9 and 163.2 eV, originating from S 2p_{1/2} and S 2p_{3/2}, are attributed to S₂²⁻ in NiS₂ nanosheets, the others at 162.4 and 161.2 eV are the S 2p_{1/2} and S 2p_{3/2} of S²⁻ for CdS nanospheres, respectively.³⁸ Significantly, the binding energies of Cd 3d characteristic peaks in NiS₂/CdS composite shifted slightly to higher values compared to pure CdS, implying that the density of electrons in Cd decreases after adding NiS₂ on CdS spheres and the existence of strong interfacial interactions between CdS nanospheres and NiS₂ nanosheets. In the core-level XPS spectrum of Ni 2p (Fig. 2i), the peak at about 855.5 eV is assigned to Ni 2p_{3/2}, and the peak at 873.2 eV is assigned to Ni 2p_{1/2}. Compared with the bare NiS₂ nanoparticles (the binding energy peaks of Ni 2p_{3/2} and Ni 2p_{1/2} are 855.8 eV and 873.5 eV, respectively), the binding energy of Ni was negatively shifted by 0.3 eV (Fig. S5†), indicating, a strong electronic interaction between NiS₂ and CdS. Two broad satellite peaks are also seen, suggesting the existence of both Ni²⁺ and Ni³⁺.³⁹

The specific surface area and pore size distribution of the prepared samples were investigated using the N₂ adsorption-desorption isotherm shown in Fig. S6a,† which exhibited a typical type IV adsorption-desorption isotherm according to IUPAC with the H3 hysteresis loop at the *P/P*₀ range of 0.5–1.0.⁴⁰ Fig. S6b† shows the pore size distribution curve estimated by the Barrett-Joyner-Halender (BJH) method. These mesopores could be due to the outcome of excessive NiS₂ nanoparticles accumulated on the surface of CdS SPs. Moreover, NiS₂/CdS nanocomposites have a larger specific BET surface area than that bare CdS SPs.

Evaluation of photocatalytic performance

The photocatalytic activities of individual CdS and NiS₂/CdS for the oxidation of HMF to DFF and H₂ (eqn (1)) were measured in water under visible light irradiation. To optimize the catalyst amount, experiments adding different amounts (10 mg, 25 mg, 50 mg, 100 mg, and 150 mg) of 10% NiS₂/CdS catalyst were conducted in 50 mL of HMF (10 mM) aqueous solution. As shown in Fig. S7a,† it is obvious that the lower the amount of sample added, the greater the catalytic rate per unit mass. However, when a smaller amount of materials was added, the total product per unit volume of the solution also

became less for a certain period of time (Fig. S7b†), which is not conducive to industrialized production (large quantities of products cannot be obtained in a short period of time). Based on the comprehensive consideration of the amount of the catalyst used and the output per unit time per unit volume of the solution, either 25 mg or 50 mg of the catalyst in 50 mL of HMF (10 mM) in water is a good choice. Herein, we added 50 mg samples in 50 mL of HMF (10 mM) aqueous solution as the unified standard to continue the next series of research.

As illustrated in Fig. 3a, the yield rates of high value-added fine chemical DFF and H₂ over bare CdS are around 51.8 and 48.6 μmol g⁻¹ h⁻¹, respectively. The selectivity of DFF is about 82.0% due to the generation of trace byproducts such as 2,5-furandimethanol (DHMF) and 5-hydroxymethyl-2-furan-carboxylic acid (HMFCFA) (Fig. S8†).⁴¹ Notably, the addition of NiS₂ considerably increases the production rates of DFF and H₂, showing that NiS₂ is an efficient way to enhance photocatalytic systems. Moreover, the production rates of DFF and H₂ evolution are also dependent on the amount of NiS₂ in NiS₂/CdS. The optimized amount of NiS₂ in NiS₂/CdS was 10%. DFF and H₂ can be produced at rates of up to 519.64 and 476.60 μmol g⁻¹ h⁻¹, respectively, which are around 10.03 and 9.81 times higher than that on bare CdS. Additionally, DFF's selectivity over 10% NiS₂/CdS could increase up to 90.2%. When the NiS₂ loading reached 12%, the generation rate decreased significantly. This is because NiS₂ and CdS could expose most interfaces on the surface of 10% NiS₂/CdS, therefore, there were most corresponding photocatalytic active sites. Intriguingly, as the proportion of NiS₂ increases, so does the selectivity of DFF with a maximum value of 95.2%. From

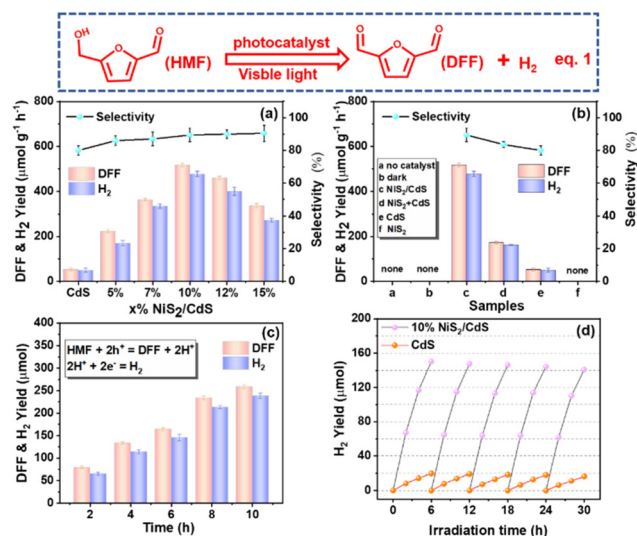


Fig. 3 (a) Photocatalytic selective transformation of HMF for DFF conversion and H₂ production and DFF selectivity over CdS and x% NiS₂/CdS samples; (b) comparison of CdS, NiS₂, 10% NiS₂/CdS, 10% NiS₂ + CdS samples and no catalyst added for the photocatalytic selective transformation of HMF; (c) time-online photocatalytic selective transformation of HMF activity over the 10% NiS₂/CdS sample. (d) Cyclic photocatalytic selective transformation of HMF over CdS and 10% NiS₂/CdS samples.

Fig. 3b, it is evident that 10% NiS₂ + CdS has significantly less photocatalytic activity than 10% NiS₂/CdS, but stronger photocatalytic activity than CdS. This indicates that NiS₂ is indeed an effective cocatalyst at enhancing the photocatalytic activity of CdS, and it is quite necessary to optimize the interfaces between the photocatalyst and the cocatalyst. The control studies, on the other hand, showed that DFF and H₂ were undetectable in the absence of photocatalysts or visible light irradiation. These findings demonstrate that: (1) the production of DFF and H₂ is actually activated by visible-light-driven photocatalysis; (2) NiS₂ as a cocatalyst could efficiently enhance the photocatalytic activity of CdS for the conversion of DFF and H₂; (3) the synergistic effect between NiS₂ and CdS is an ideal prerequisite for enhancing the photoredox activity. The reaction time was extended from 2 hours to 10 hours to examine the catalyst performance over time (Fig. 3c). No peaks for other possible products were detected, indicating the high purity of the photocatalytic biomass products (Fig. S9†). It can be seen that the photocatalytic activity of 10% NiS₂/CdS on the generation of DFF and H₂ increases gradually with the increase of the reaction time. The reactive stability of 10% NiS₂/CdS as a representative sample has been evaluated by examining the H₂ yield using the same system for photocatalytic conversion, repeatedly, five times. As shown in Fig. 3d, the decay for long-time visible-light-driven hydrogen evolution of pure CdS is larger than that of 10% NiS₂/CdS as a result of the known photocorrosion of metal sulfides. The binding energy changes of Cd in CdS and 10% NiS₂/CdS before and after the photostability test can be seen in Fig. S10.† The results show that the cocatalyst NiS₂ can inhibit the photocorrosion of CdS. The photostability of NiS₂/CdS composites is related to the effective separation of the photoexcited electrons and holes in CdS by NiS₂. The crystallinity (Fig. S11†) and morphology (Fig. S12†) of the material remained unchanged after irradiation, indicating that the photoreaction had no effect on the morphology and crystal morphology of the material. The crystallinity and morphology of NiS₂/CdS nanospheres do not change after the photoreaction, which indicates that the photoreaction did not affect the morphology and crystal form of the material. We compared the photocatalytic oxidation of HMF with those of the previously reported photocatalysts under both similar and different conditions (Table S1†). We can see that the photocatalyst in our work clearly exhibits a high photocatalytic rate and high selectivity.

Furfuryl alcohol (biomass derivative) as an analog of HMF can be converted to valuable platform chemicals through high-temperature catalysis in the solution.⁴² To check the broad scope, we selected furfural alcohol for photocatalytic conversion on 10% NiS₂/CdS nanocrystals. Furfural alcohol (one of the biomass derivatives) has been converted to valuable platform chemicals biomass derivatives through high-temperature catalysis in the solution. The catalyst performance over time is shown in Fig. S13† (the corresponding standard curves are shown in Fig. S14†). Furfural and H₂ can be produced at rates of up to 611.40 and 561.86 μmol g⁻¹ h⁻¹, respectively. The selectivity of furfural was about 98.7%.

Photocatalytic mechanism

The optical characteristics of pure CdS and its mixtures were also investigated by the UV-vis diffuse reflectance spectra. As shown in Fig. S15a,† the spectrum shifted upwards with increasing NiS₂ content. Similar to pure CdS nanosheets, NiS₂/CdS showed typical semiconducting absorption at around 520 nm. NiS₂ loaded on CdS nanospheres demonstrated a considerable increase in light absorption from 500 to 750 nm due to its strong light absorption ability. In addition, the loading of NiS₂ does not affect the light absorption edges, indicating that it does not influence the lattice of CdS. Fig. S15b† shows the band gaps of CdS, 10% NiS₂/CdS, and 15% NiS₂/CdS. Band gap energy (E_g) values for other samples can be found in Table S2.† The band gap energy values of CdS, 5% NiS₂/CdS, 7% NiS₂/CdS, 10% NiS₂/CdS, 12% NiS₂/CdS, and 15% NiS₂/CdS were 2.43 eV, 2.42 eV, 2.41 eV, 2.39 eV, 2.38 eV, and 2.37 eV, respectively. With the increasing content of NiS₂, the band gap energy values of the samples decrease, which indicates that the decoration of the CdS lattice with NiS₂ slightly decreases its band gap. The band gap energy values of the products are evaluated using the Kubelka–Munk function, and the corresponding formula is as follows:

$$(ahv)^2 = c(hv - E_g) \quad (3)$$

where a and v represent the absorption coefficient and frequency, respectively, c and h are constants.

Capacitance measurements by the Mott–Schottky equation are used to quantify electron carrier density.⁴³

$$\frac{1}{C^2} = \left(\frac{2}{N_D e_0 \epsilon \epsilon_0} \right) \left[(E - E_{FB}) - \frac{kT}{e_0} \right] \quad (4)$$

where C is the space charge capacity of the semiconductor, N_D refers to the electron carrier density, ϵ_0 and ϵ are set as the permittivity of a vacuum and the relative permittivity of the semiconductor, E , e_0 , E_{FB} , k , and T are the applied potential, elemental charge value, flat band potential, Boltzmann constant, and temperature, respectively. As shown in Fig. S16,† the slope of the Mott–Schottky plots and n-type characteristics of both pure CdS and 10% NiS₂/CdS can be obtained. The experimentally determined flat-band potentials (E_{FB}) of bare CdS and 10% NiS₂/CdS are -0.71 V and -0.63 V (vs. Ag/AgCl), respectively. Assuming that E_{FB} is equivalent to the conduction band (E_{CB}) position, the band gap energy E_g of CdS and 10% NiS₂/CdS are 2.43 eV and 2.39 eV, respectively. According to the formula $E_g = E_{VB} - E_{CB}$, the valence band edge potentials for bare CdS and 10% NiS₂/CdS is 1.72 V and 1.76 V, respectively. For the conversion of HMF into DFF, the potential of HMF to DFF ($E_{HMF/DFF}$) and the potential of DFF to oxidized DFF ($E_{DFF/oxidized DFF}$) are 1.61 V and 2.03 V, respectively.⁴⁴ The VB of CdS is exactly halfway between these two potentials, resulting in the major product of DFF (Scheme 1).

The charge transfer mechanism at the NiS₂/CdS interface was investigated using the density functional theory (DFT). The theoretical work functions (Φ) of CdS (Fig. 4a) and NiS₂ (Fig. 4b) were calculated to be 6.46 and 8.76 eV, respectively,



Scheme 1 Schematic illustration of the determined CB and VB edges of NiS₂/CdS and oxidizing potential.

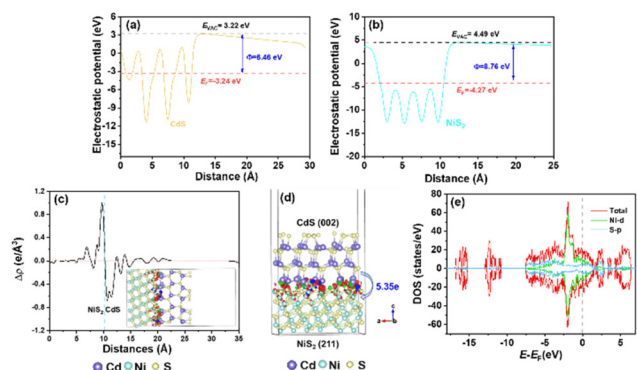
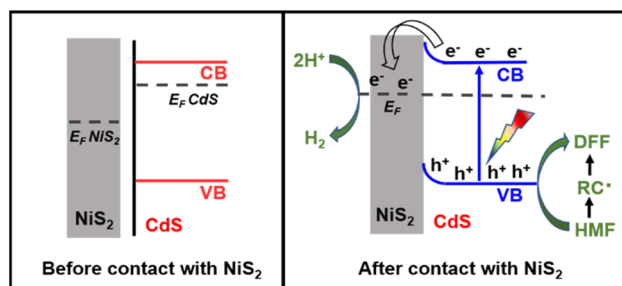


Fig. 4 Theoretical work functions of CdS (a) and NiS₂ (b). (c) Electron density difference (e Å⁻³) and (d) Bader electron transfer at the NiS₂/CdS interfaces. (e) The density of states of NiS₂.

indicating that the electrons are prone to gather on NiS₂. The deformation charge density (Fig. 4c) shows that electrons tend to transfer from CdS to NiS₂ at the NiS₂/CdS interface. Therefore, the S atoms in NiS₂ primarily facilitated the gain of electrons. Specifically, the Bader charge shows that the electron transport from CdS to NiS₂ is 5.35e (Fig. 4d). The electron transfer route is consistent with the binding energy shifts of S 2p and the obtained work function above.

According to the aforementioned findings, a possible catalytic mechanism for the photocatalytic conversion of HMF into DFF along with H₂ evolution on NiS₂/CdS nanospheres is proposed (Scheme 2). NiS₂ has metallic properties and high electrical conductivity. The density of states also confirmed the metallic nature of NiS₂ (Fig. 4e). An interfacial barrier area, also known as a barrier layer, forms at the contact interfaces of NiS₂ and CdS when they are in close proximity. Due to the lower Fermi level (E_F NiS₂) compared to that of CdS (E_F CdS) and surface metallic characteristics, the photogenerated electrons from CdS can quickly move to NiS₂, and the recombination of photogenerated electron-hole pairs is prevented. Additionally, NiS₂ can easily mix with surface-adsorbed H₂O to



Scheme 2 Energy band diagrams of the pure CdS and NiS₂/CdS before contact with metallic NiS₂ and the interfacial photoinduced charge transfer and reaction after contact with NiS₂ and light irradiation.

generate Ni-H bonds in addition to serving as active sites for the hydrogen evolution reaction. The electrons on H₂O-bonded NiS₂ will generate H₂ with adsorbed H⁺, which further enhances the photocatalytic performance and the NiS₂/CdS nanohybrid system stability. The photoexcited electrons travel to the CB of the primary catalyst CdS under visible light, while an equivalent number of photoexcited holes are left on the VB of CdS. The cocatalyst NiS₂ receives the electrons quickly delivered from the CB of CdS. In order to convert HMF into DFF, photoexcited holes on CdS first activate the alkoxide anion that was deprotonated from HMF, resulting in the formation of appropriate carbon radicals and the release of an H⁺ ion. The generated carbon radicals then continue to react with the holes to create DFF molecules. One HMF molecule can produce two H⁺ ions in total, which can combine with two electrons on NiS₂. Therefore, the two isolated reactions can be successfully carried out without the use of any sacrificial agents, and aqueous solution green recycling can be developed.

Photoluminescence (PL) spectra and photoelectric properties

To further investigate the effect of the deposited NiS₂ in separating the internal carriers on the photocatalytic coproduction of H₂ and DFF, the PL spectra of pure CdS and 10% NiS₂/CdS were measured. The PL emission intensity was clearly quenched when the NiS₂ was modified on CdS (Fig. 5a). The time-resolved PL (TRPL) decay spectra were also recorded to explore the carrier transfer dynamics, as shown in Fig. 5b. The decay curves of the produced samples match the following double exponential equation quite well:

$$I(t) = I_0 + A_1 \exp\left(\frac{-t}{\tau_1}\right) + A_2 \exp\left(\frac{-t}{\tau_2}\right) \quad (5)$$

where I_0 is the baseline correction value; A_1 and A_2 are the pre-factors, respectively; τ_1 and τ_2 are the lifetimes (ns) of the radiative and nonradiative recombination processes, respectively.⁴⁵ The calculation formula of the average life (τ_{ave}) is as follows:

$$\tau_{ave} = (A_1\tau_1^2 + A_2\tau_2^2)/(A_1\tau_1 + A_2\tau_2) \quad (6)$$

The double-exponential fitting results are summarized in Table S3.† A significant PL quenching and lifespan enhance-



Fig. 5 (a) PL spectra, (b) TRPL decay spectra, (c) transient photocurrent responses, and (d) EIS spectra for pure CdS and 10% NiS₂/CdS; (e) schematic of photochemical cell and electron flow; (f) *in situ* EPR spectra for CdS, NiS₂ and 10% NiS₂/CdS in a mixture solution of HMF in the presence of spin-trapping agent DMPO with or without visible light irradiation.

ment suggested that carrier transfer and separation in NiS₂/CdS were significantly more efficient.⁴⁶ Photoelectrochemical tests were also carried out to further examine the separation and transfer ability of the charges. The transient photocurrent responses of the two samples exposed to visible light (420 nm) with regular on/off cycles are depicted in Fig. 5c. It is evident that the 10% NiS₂/CdS sample displayed much greater photocurrent density than that of CdS, implying a more efficient carrier separation in 10% NiS₂/CdS.⁴⁷ Moreover, the electrochemical impedance spectroscopy (EIS) Nyquist plot of 10% NiS₂/CdS exhibited a smaller arc radius than pure CdS (Fig. 5d), which suggested that there was a lower interface charge transfer resistance in 10% NiS₂/CdS.

Fig. S17† shows the LSV polarization curves of pristine CdS, NiS₂, and 10% NiS₂/CdS at a scan rate of 5 mV s⁻¹. It is clear to see that the electrocatalytic activity is NiS₂ > 10% NiS₂/CdS > CdS. This indicates that NiS₂ can effectively accelerate the electrocatalytic hydrogen evolution kinetics of nanocomposites, thereby, greatly improving the HER (hydrogen evolution reaction) performance of nanocomposites.⁴⁸ The corresponding schematic of the photochemical cell and electron flow is shown in Fig. 5e. Nevertheless, all of the above results prove that the NiS₂/CdS nanosphere structure could accelerate the photogenerated carrier separation, thus improving the photocatalytic performance.

Electron paramagnetic resonance study

In situ EPR studies were performed to detect reactive intermediates during the catalytic reactions and understand the photocatalytic dehydrogenation process of HMF in prepared

composites. As illustrated in Fig. 5f, no EPR signals could be seen on the blank tests (performed in the dark or under visible light irradiation with bare NiS₂). But after irradiation, signals corresponding to the peaks of carbon-centered radicals were clearly seen across both pure CdS and 10% NiS₂/CdS. The intensity of the signals increased with the radiation time, indicating the formation of carbon-centered radicals over 10% NiS₂/CdS during the photocatalytic reaction.⁴⁹ Moreover, the intensity of 10% NiS₂/CdS is clearly stronger than that of CdS, showing that more carbon-centered radicals are produced over 10% NiS₂/CdS. The results are consistent with the above photocatalytic activity.

Conclusions

In this study, NiS₂/CdS was synthesized *via* a one-step deep eutectic solvothermal procedure. The obtained NiS₂/CdS composites, especially 10% NiS₂/CdS nanospheres, can be used as a highly effective, bifunctional photocatalyst for the selective conversion of biomass-derived alcohols to high-value-added fine chemical DFF and clean energy (H₂) without adding the sacrificial agent. DFF and H₂ can be produced at rates of up to 519.64 and 476.60 μmol g⁻¹ h⁻¹, which are about 10.03 and 9.81 times larger, respectively, than that on bare CdS under visible light irradiation. The DFF selectivity of x% NiS₂/CdS can reach up to 95.2%. Photocurrent, EIS, LSV, PL, and TRPL experiments and DFT calculations clearly showed that NiS₂ can effectively facilitate the separation and migration of photo-excited charges when used as a co-catalyst. A rational photo-

catalytic process was proposed on the basis of UV-vis, Mott-Schottky, and EPR. Importantly, this work offers a promising method for creating innovative, non-precious metal, and efficient visible-light-driven composite photocatalysts. The photocatalytic dehydrogenation strategy could be used in other organic transformation processes.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- S. Xu, P. Zhou, Z. Zhang, C. Yang, B. Zhang, K. Deng, S. Bottle and H. Zhu, *J. Am. Chem. Soc.*, 2017, **139**, 14775–14782.
- T. Chhabra, P. Dwivedi and V. Krishnan, *Green Chem.*, 2022, **24**, 898–910.
- T. Chhabra, A. Bahuguna, S. S. Dhankhar, C. M. Nagaraja and V. Krishnan, *Green Chem.*, 2019, **21**, 6012–6026.
- C. Xu, E. Paone, D. Rodriguez-Padron, R. Luque and F. Mauriello, *Chem. Soc. Rev.*, 2020, **49**, 4273–4306.
- Y. C. Yang and T. C. Mu, *Green Chem.*, 2021, **23**, 4228–4254.
- T. Wang, L. Tao, X. Zhu, C. Chen, W. Chen, S. Du, Y. Zhou, B. Zhou, D. Wang, C. Xie, P. Long, W. Li, Y. Wang, R. Chen, Y. Zou, X.-Z. Fu, Y. Li, X. Duan and S. Wang, *Nat. Catal.*, 2022, **5**, 66–73.
- K. Gu, D. Wang, C. Xie, T. Wang, G. Huang, Y. Liu, Y. Zou, L. Tao and S. Wang, *Angew. Chem., Int. Ed.*, 2021, **60**, 20253–20258.
- C. Li, J. Li, L. Qin, P. Yang and D. G. Vlachos, *ACS Catal.*, 2021, **11**, 11336–11359.
- D. A. Giannakoudakis, V. Nair, A. Khan, E. A. Deliyanni, J. C. Colmenares and K. S. Triantafyllidis, *Appl. Catal., B*, 2019, **256**, 117803.
- A. Khan, M. Goepel, A. Kubas, D. Lomot, W. Lisowski, D. Lisovytskiy, A. Nowicka, J. C. Colmenares and R. Glaser, *ChemSusChem*, 2021, **14**, 1351–1362.
- Z. H. Zhang and K. J. Deng, *ACS Catal.*, 2015, **5**, 6529–6544.
- M. Zhang, Z. Li, X. Xin, J. Zhang, Y. Feng and H. Lv, *ACS Catal.*, 2020, **10**, 14793–14800.
- X. Huang, O. Akdim, M. Douthwaite, K. Wang, L. Zhao, R. J. Lewis, S. Pattison, I. T. Daniel, P. J. Miedzkiak, G. Shaw, D. J. Morgan, S. M. Althahban, T. E. Davies, Q. He, F. Wang, J. Fu, D. Bethell, S. McIntosh, C. J. Kiely and G. J. Hutchings, *Nature*, 2022, **603**, 271–275.
- E. Hayashi, Y. Yamaguchi, K. Kamata, N. Tsunoda, Y. Kumagai, F. Oba and M. Hara, *J. Am. Chem. Soc.*, 2019, **141**, 890–900.
- L. Hu, A. Y. He, X. Y. Liu, J. Xia, J. X. Xu, S. Y. Zhou and J. M. Xu, *ACS Sustainable Chem. Eng.*, 2018, **6**, 15915–15935.
- X. Wu, N. Luo, S. Xie, H. Zhang, Q. Zhang, F. Wang and Y. Wang, *Chem. Soc. Rev.*, 2020, **49**, 6198–6223.
- T. Chhabra, S. Dhingra, C. M. Nagaraja and V. Krishnan, *Carbon*, 2021, **183**, 984–998.
- S. Dhingra, T. Chhabra, V. Krishnan and C. M. Nagaraja, *ACS Appl. Energy Mater.*, 2020, **3**, 7138–7148.
- S. Yurdakal, B. S. Tek, O. Alagoz, V. Augugliaro, V. Loddo, G. Palmisano and L. Palmisano, *ACS Sustainable Chem. Eng.*, 2013, **1**, 456–461.
- H. Zhang, Q. Wu, C. Guo, Y. Wu and T. Wu, *ACS Sustainable Chem. Eng.*, 2017, **5**, 3517–3523.
- M. D. Regulacio and M. Y. Han, *Acc. Chem. Res.*, 2016, **49**, 511–519.
- P. Zhang, S. B. Wang, B. Y. Guan and X. W. Lou, *Energy Environ. Sci.*, 2019, **12**, 164–168.
- X. Wu, S. Xie, H. Zhang, Q. Zhang, B. F. Sels and Y. Wang, *Adv. Mater.*, 2021, **33**, 2007129.
- S. Dhingra, M. Sharma, V. Krishnan and C. M. Nagaraja, *J. Colloid Interface Sci.*, 2022, **608**, 1040–1050.
- Y. Zhong, G. Zhao, F. Ma, Y. Wu and X. Hao, *Appl. Catal., B*, 2016, **199**, 466–472.
- X.-j. Wang, X.-l. Li, C. Liu, F.-t. Li, Y.-p. Li, J. Zhao, R.-h. Liu and G.-d. Li, *Int. J. Hydrogen Energy*, 2018, **43**, 219–228.
- B. He, C. Bie, X. Fei, B. Cheng, J. Yu, W. Ho, A. A. Al-Ghamdi and S. Wageh, *Appl. Catal., B*, 2021, **288**, 119994.
- Z. Dou, Z. Zhang, H. Zhou and M. Wang, *Angew. Chem., Int. Ed.*, 2021, **60**, 16399–16403.
- Y. Wang, Z. Fang, Q. Dong, Y. Chu, X. Shi, M. Song and Z. Hao, *Appl. Surf. Sci.*, 2019, **491**, 590–594.
- D. Yu, Z. Xue and T. Mu, *Cell Rep. Phys. Sci.*, 2022, 100809.
- D. Yu, Z. Xue and T. Mu, *Chem. Soc. Rev.*, 2021, **50**, 9345–9345.
- X. Zhao, X. Lan, D. Yu, H. Fu, Z. Liu and T. Mu, *Chem. Commun.*, 2018, **54**, 13010–13013.
- G. Kresse and J. Furthmuller, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1996, **54**, 11169–11186.
- J. Yuan, J. Wen, Y. Zhong, X. Li, Y. Fang, S. Zhang and W. Liu, *J. Mater. Chem. A*, 2015, **3**, 18244–18255.
- J. Yue, J. Xu, J. Niu and M. Chen, *Int. J. Hydrogen Energy*, 2019, **44**, 19603–19613.
- J. Zou, W. Zhou, L. Huang, B. Guo, C. Yang, Y. Hou, J. Zhang and L. Wu, *J. Catal.*, 2021, **400**, 347–354.
- B. Zhang, H. Fu and T. Mu, *Green Chem.*, 2022, **24**, 877–884.
- J. Zou, W. Zhou, L. Huang, B. Guo, C. Yang, Y. Hou, J. Zhang and L. Wu, *J. Catal.*, 2021, **400**, 347–354.
- J. Liang, Y. Wang, Q. Liu, Y. Luo, T. Li, H. Zhao, S. Lu, F. Zhang, A. M. Asiri, F. Liu, D. Ma and X. Sun, *J. Mater. Chem. A*, 2021, **9**, 6117–6122.
- K. S. W. Sing, *Pure Appl. Chem.*, 1985, **57**, 603–619.
- S. Dhingra, M. Sharma, V. Krishnan and C. M. Nagaraja, *J. Colloid Interface Sci.*, 2022, **615**, 346–356.

- 42 T. Chhabra, J. Rohilla and V. Krishnan, *Mol. Catal.*, 2022, **519**, 112135.
- 43 L. Shi, W. Zhou, Z. Li, S. Koul, A. Kushima and Y. Yang, *ACS Nano*, 2018, **12**, 6335–6342.
- 44 S. Meng, H. Wu, Y. Cui, X. Zheng, H. Wang, S. Chen, Y. Wang and X. Fu, *Appl. Catal., B*, 2020, **266**, 118617.
- 45 X. W. Shi, S. Kim, M. Fujitsuka and T. Majima, *Appl. Catal., B*, 2019, **254**, 594–600.
- 46 X. L. Bao, H. L. Li, Z. Y. Wang, F. X. Tong, M. Liu, Z. K. Zheng, P. Wang, H. F. Cheng, Y. Y. Liu, Y. Dai, Y. C. Fan, Z. Y. Li and B. B. Huang, *Appl. Catal., B*, 2021, **286**, 119885.
- 47 M. Liu, L. R. Zheng, X. L. Bao, Z. Y. Wang, P. Wang, Y. Y. Liu, H. F. Cheng, Y. Dai, B. B. Huang and Z. K. Zheng, *Chem. Eng. J.*, 2021, **405**, 126654.
- 48 C. Li, H. Wang, S. B. Naghadeh, J. Z. Zhang and P. Fang, *Appl. Catal., B*, 2018, **227**, 229–239.
- 49 X. Bao, M. Liu, Z. Wang, D. Dai, P. Wang, H. Cheng, Y. Liu, Z. Zheng, Y. Dai and B. Huang, *ACS Catal.*, 2022, **12**, 1919–1929.