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## Controlling electrocatalytic nitrate reduction efficiency by utilizing $d\pi-p\pi$ interactions in parallel stacking molecular systems†

Sourav Bhowmick, <sup>ad</sup> Ashadul Adalder, <sup>a</sup> Abhishek Maiti, <sup>b</sup> Samadhan Kapse, <sup>c</sup> Ranjit Thapa, <sup>c</sup> Supriya Mondal <sup>d</sup> and Uttam Kumar Ghorai <sup>\*a</sup>

Electrochemical reduction of nitrate to ammonia using electrocatalysts is a promising alternative strategy for both wastewater treatment and production of green ammonia. Numerous tactics have been developed to increase the electrocatalyst's  $\text{NO}_3\text{RR}$  activity. Herein, we report a unique molecular alignment-dependent  $\text{NO}_3\text{RR}$  performance using  $\alpha$ -CuPc and  $\beta$ -CuPc nanostructures as effective electrocatalysts for the ambient synthesis of ammonia. The well-aligned  $\beta$ -CuPc demonstrated an impressive ammonia yield rate of  $62\,703\, \mu\text{g h}^{-1}\, \text{mg}_{\text{cat}}^{-1}$  and a Faradaic efficiency of 96%. In contrast, the less well-aligned  $\alpha$ -CuPc exhibited a yield rate of  $36\,889\, \mu\text{g h}^{-1}\, \text{mg}_{\text{cat}}^{-1}$  and a Faradaic efficiency of 61% at  $-1.1\, \text{V}$  vs. RHE under the same conditions. Scanning tunneling microscopy/spectroscopy (STM/S) confirms that the well-aligned  $\beta$ -CuPc exhibits superior transport properties due to optimal interaction of the Cu atom with the nitrogen atom of parallel molecules ( $d\pi-p\pi$ ) in its one-dimensional nanostructure, which is clearly reflected in the electrocatalytic performance. Furthermore, theoretical research reveals that the  $\text{NO}_3\text{RR}$  is the predominant process on the  $\beta$ -CuPc catalyst in comparison to the hydrogen evolution reaction, which is verified by gas chromatography, with  $\beta$ -CuPc exhibiting weaker binding of the  $^*\text{NO}$  intermediate at the copper site and a lower overpotential, hence facilitating the  $\text{NO}_3\text{RR}$  relative to  $\alpha$ -CuPc.

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## Introduction

Ammonia ( $\text{NH}_3$ ) has a high weight hydrogen storage capacity of 17.6% and a high energy density, making it a promising carbon-free energy carrier and a medium for the next generation of renewable energy storage.<sup>1</sup> The electrocatalytic nitrate reduction reaction ( $\text{NO}_3\text{RR}$ ) is a greener substitute for the energy-intensive and ecologically polluting Haber–Bosch method for ammonia production.<sup>2–4</sup> As a result, nitrate to ammonia conversion using renewable energy may accomplish both nitrate removal and ammonia synthesis, which is very relevant for balancing sustainability issues with benefits for the economy.<sup>5,6</sup> The

electrocatalytic reduction of nitrate to ammonia is a complex procedure that requires the transfer of eight electrons and nine protons, which results in a low ammonia production rate and Faradaic efficiency. This process is kinetically slow and impeded by the competing hydrogen evolution reaction.<sup>7,8</sup> Consequently, it is imperative to identify effective catalysts to enhance the kinetics of the  $\text{NO}_3\text{RR}$ . Diverse strategies have been investigated to enhance the efficacy of the electrocatalyst by improving its intrinsic reaction activity and selectivity for the reduction of nitrate to  $\text{NH}_3$ . These strategies include defect engineering,<sup>9,10</sup> alloying,<sup>11,12</sup> the effect of dissolved oxygen,<sup>13</sup> core–shell structures,<sup>14</sup> crystal phase engineering,<sup>15</sup> strain engineering,<sup>16</sup> Janus structures,<sup>17</sup> interface engineering,<sup>18</sup> and oxygen vacancies<sup>19</sup> and many more.<sup>20,21</sup> Despite these advances, molecular alignment in nanostructure based electrocatalysts, which is one of the important parameters for selective nitrate-to-ammonia reduction, is rarely reported.

Transition metal phthalocyanine-based molecular structures have attracted considerable scientific attention because of their significant contributions to several catalytic applications, including photocatalysts,<sup>22</sup> electrocatalytic oxidation,<sup>23,24</sup> reduction processes,<sup>25–30</sup> and others. Versatility in the activity of these metal complexes and their performance depends on the metal–ligand interaction, which depends on the alignment of the molecules within the crystal. Metal phthalocyanines are

<sup>a</sup>Department of Industrial Chemistry & Applied Chemistry, Swami Vivekananda Research Centre, Ramakrishna Mission Vidyamandira, Belur Math, Howrah 711202, India. E-mail: [uttam.indchem@vidyamandira.ac.in](mailto:uttam.indchem@vidyamandira.ac.in)

<sup>b</sup>School of Physical Sciences, Indian Association for the Cultivation of Science, Jadavpur, Kolkata, 700032, India

<sup>c</sup>Department of Physics and Centre for Computational and Integrative Sciences, SRM University-AP, Amaravati 522240, Andhra Pradesh, India

<sup>d</sup>Department of Physics, Government General Degree College Chapra, Nadia, West Bengal, 741123, India

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‡ Present address: Laboratoire de Physique des Solides, Université Paris-Saclay, CNRS, Orsay, 91405, France.



combinations of porphyrin derivatives with a core metallic atom attached to a  $\pi$ -conjugated ligand and aromatic molecules that may be stacked by  $\pi$ - $\pi$  supramolecular interactions. The characteristics of these compounds are determined by their structure. The  $\pi$ - $\pi$  stacking along with the aromaticity of phthalocyanine molecules provides excellent thermal and chemical stability. Due to the planar macrocyclic structure and the availability of  $\pi$  electrons, 1D growth of phthalocyanines is favourable through  $\pi$ - $\pi$  interactions. Their excellent thermal stability, good electron response, flexibility and easy fabrication technique have endowed them with multifunctional applications.<sup>31</sup> Most of the metal phthalocyanines exist in either triclinic or monoclinic polymorphism.<sup>32</sup> The polymorphs of metal phthalocyanines depend on their arrangement of molecular macrocycles and the substrate surface–molecular interactions.<sup>33</sup> Hence, charge carrier mobility and conductivity through  $\pi$ - $\pi$  interactions in one-dimensional nanostructures, depending on molecular orientation, may influence the electrocatalytic performance. Among the metal phthalocyanine nanostructures, CuPc is a widely used molecule for fundamental and technological applications due to its semiconductor nature.<sup>31</sup> CuPc crystallizes into four phases  $\alpha$ ,  $\beta$ ,  $\gamma$ , and  $\delta$  and  $\alpha$  and  $\beta$  are the most stable phases. Within the herringbone structure of the two  $\alpha$  and  $\beta$  phases, the molecules arrange themselves in stacks interconnected by van der Waals interactions. These stacks are oriented at angles of 26.5° and 45.8°, respectively, between the column direction (*b*-axis) and the molecular plane.<sup>34</sup> Recently, Dutt *et al.*<sup>35</sup> studied the geometry dependent electrocatalytic oxygen reduction reaction and hydrogen evolution properties using the regio-isomers of tetra-

amino-substituted cobalt phthalocyanine. To our knowledge, there has yet to be a study focused on the molecular alignment dependent electrocatalyst nitrate reduction reaction.

We have fabricated  $\alpha$ -CuPc and  $\beta$ -CuPc electrocatalysts, with  $\beta$ -CuPc exhibiting better alignment compared to  $\alpha$ -CuPc and studied the electrocatalytic  $\text{NO}_3\text{RR}$ . To understand the effect of the molecular alignment on the electrical conductivities for both the  $\alpha$ -CuPc and  $\beta$ -CuPc, STM measurements were carried out, suggesting that the  $\beta$ -phase should have better transport properties compared to its  $\alpha$  counterpart. We then investigated the impact of molecular alignment on the electrocatalytic performance of the nitrate reduction reaction ( $\text{NO}_3\text{RR}$ ). The sandwiched structured  $\beta$ -CuPc showed an ammonia yield rate of  $62\,703\, \mu\text{g h}^{-1}\, \text{mg}_{\text{cat}}^{-1}$  and 96% Faradaic efficiency, whereas less well aligned  $\alpha$ -CuPc showed a yield rate of  $36\,889\, \mu\text{g h}^{-1}\, \text{mg}_{\text{cat}}^{-1}$  and 61% Faradaic efficiency at  $-1.1\, \text{V}$  *vs.* RHE under ambient conditions. The overpotential ( $\eta$ ) values at the Cu site of  $\alpha$ -CuPc and  $\beta$ -CuPc are 0.72 eV and 0.48 eV, respectively, mostly dependent on the adsorption of the  $^*\text{NO}$  intermediate. Consequently,  $\beta$ -CuPc demonstrates enhanced catalytic efficacy for the  $\text{NO}_3\text{RR}$ .

## Results and discussion

### STM characterization for predicting electrocatalytic efficacy

The electrocatalytic performance of a material is strongly governed by its intrinsic electrical transport properties. Hence, before proceeding to the study of the molecular orientation dependent electrocatalysis properties of  $\alpha$ - and  $\beta$ -phthalocyanines, we investigated their electrical transport properties, both

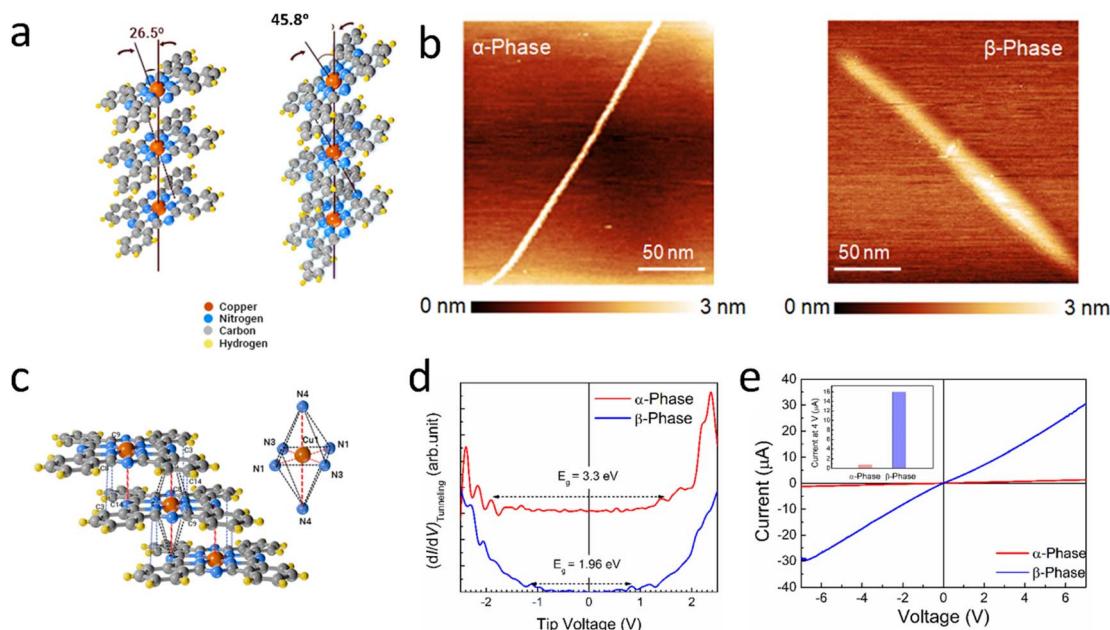


Fig. 1 (a) Schematic diagram of  $\alpha$ -CuPc and  $\beta$ -CuPc. (b) STM topographies (recorded in constant current mode) of  $\alpha$ - and  $\beta$ -phthalocyanines. (c) Schematic representation of intermolecular interaction in  $\beta$ -CuPc. (d) Differential tunnel conductance ( $dI/dV$ )<sub>Tunneling</sub> spectra of  $\alpha$ - and  $\beta$ -phthalocyanines to understand their transport gap. (e) Current–voltage characteristics ( $I$ – $V$ ) of bulk  $\alpha$ - and  $\beta$ -phthalocyanines as obtained from two-probe measurements. Inset shows a comparison of current at 4 V. All the results infer better transport properties of  $\beta$ -phthalocyanines.



at the nanoscale and in the bulk of the materials. In this regard, we first characterized the phthalocyanines through a localized mode of measurements namely scanning tunneling microscopy (STM). For such measurements, synthesized phthalocyanine samples dispersed in ethanol with vigorous sonication were drop-cast and dried on a freshly cleaved highly oriented pyrolytic graphite (HOPG) surface. In Fig. 1b, the STM topography of the fraction of phthalocyanines recorded in constant current mode has been presented. Although the diameter of the fraction of the  $\beta$ -phase was found to be higher than that of the  $\alpha$ -phase, the results infer that both phthalocyanines have a nanowire kind of shape. We then proceeded to record the tunneling current *vs.* voltage ( $I_{\text{Tunneling}}-V$ ) of the phthalocyanines at room temperature by positioning the tip over an individual phthalocyanine nanowire and momentarily disabling the feedback loop. Although a comparison of current at a particular voltage could provide a preliminary idea on the conductivities of  $\alpha$ - and  $\beta$ -phthalocyanines, such analysis is not possible from STS, as

the same tip–sample separation could not be ensured for both compounds. In this direction, to understand the transport properties, analysis of the transport gap would be a better approach. We could locate the band edges and transport gap of the phthalocyanines from STS as the differential tunnel conductance  $\{(dI/dV)_{\text{Tunneling}}\}$  spectra reflect the density of states,  $(dI/dV)_{\text{Tunneling}} \propto \text{DOS}$ . Here, the work functions of the substrate electrode and tip are aligned at 0 V, and the band energies are determined with regard to Fermi energy (EF). The conduction band (CB) edge was shown by the first peak in the negative voltage region, which was the point at which electrons could be injected from the tip to the semiconductor, because the bias was provided to the tip with regard to the substrate electrode. In a similar vein, the valence band (VB) edge, which marks the point at which electrons can be extracted from phthalocyanines, was indicated by the first peak at the positive voltage. We observed a characteristically higher transport gap of the  $\alpha$ -phase, resulting in inferior transport properties of the

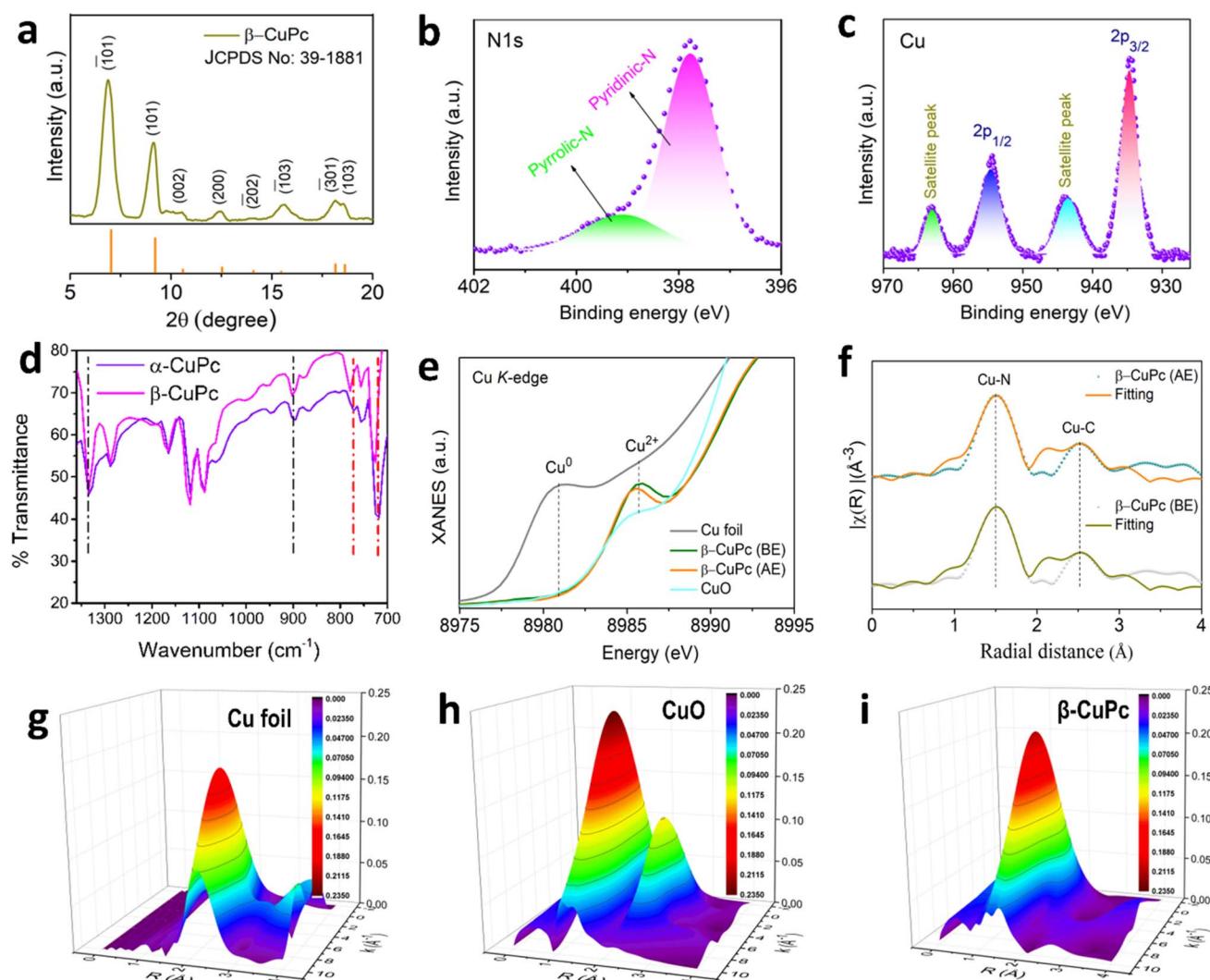


Fig. 2 (a) XRD plot of  $\beta$ -CuPc, (b) high resolution XPS N 1s spectra of  $\beta$ -CuPc, (c) high resolution XPS Cu 2p spectra of  $\beta$ -CuPc, (d) FTIR spectra of  $\alpha$ -CuPc and  $\beta$ -CuPc, (e) Cu K-edge XANES spectra of  $\beta$ -CuPc before and after electrolysis, (f) EXAFS spectra of  $\beta$ -CuPc before and after electrolysis, and (g–i) wavelet-transform EXAFS plot of Cu foil, CuO and  $\beta$ -CuPc.



compound compared to the  $\beta$ -phase (Fig. 1d). It may be noted that a larger transport gap reflects higher carrier effective mass, leading to inferior transport due to the presence of fewer charge carriers available for conduction. Hence, a comparison of transport gaps suggests that the  $\beta$ -phase should have better transport properties compared to its  $\alpha$  counterpart.

To further validate these results, we carried out current-voltage ( $I$ - $V$ ) measurements in their bulk through a conventional two probe measurement. A higher current was witnessed in the  $\beta$ -phase (Fig. 1e) to conclusively state that the  $\beta$ -phase has better electrical transport properties. It would be anticipated that the unpaired 3d electrons of Cu will interact with the ligating nitrogen atoms at  $\sim 1.83$  Å. The molecule contains four additional nitrogen atoms spaced  $\sim 3.38$  Å apart in the same molecular structure.

In  $\beta$ -CuPc, the central Cu atom lies exactly over and under the nitrogen molecules of parallel molecules due to its angle of  $45.8^\circ$  with the  $b$  axis (Fig. 1a). The closest distance of parallel  $\beta$ -CuPc is  $\sim 3.38$  Å, which is the distance between intermolecular Cu and N (N4) (Fig. 1c). It is possible to think of the central Cu atom as being at the heart of a square-bonded structure inside a wider octahedral array of nitrogen atoms. It can also be considered a sandwiched structure with the Cu atom in the centre and nitrogen of the parallel molecule above and below Cu. But it may be deduced that the extra molecular nitrogen in the closest molecules does not form the complete nitrogen octahedron in the  $\alpha$ -phase.<sup>36</sup> These interactions of central Cu with nitrogen of the parallel molecule may play a vital role in enhancing the electrical transport of  $\beta$ -CuPc.

### Structural characterization of $\alpha$ -CuPc and $\beta$ -CuPc

X-ray diffraction (XRD) analysis was employed to assess the phase purity of the synthesized  $\beta$ -CuPc. The resulting XRD plot exhibits two prominent peaks at  $6.9^\circ$  and  $9.1^\circ$ , corresponding to the (101) and (101) planes of  $\beta$ -CuPc, respectively (Fig. 2a). Furthermore, all other minor peaks observed in the pattern are consistent with the reference pattern of  $\beta$ -CuPc (JCPDS data card no: 39-1881), confirming the successful formation of the desired  $\beta$ -CuPc phase. The formation of  $\alpha$ -CuPc was confirmed by matching XRD with JCPDS card no 36-1883 (Fig. S1a†). X-ray photoelectron spectroscopy (XPS) was utilized to examine the elemental composition and the chemical state of atoms within the CuPc molecule. The survey spectrum in Fig. S2† confirms the presence of Cu, N, and C elements in CuPc. Notably, the N 1s peak can be deconvoluted into two distinct components (Fig. 2b), corresponding to the nitrogen atoms in different chemical environments: the iso-indole nitrogen bonded to the central copper atom at 399.1 eV and the azomethine nitrogen bridging the carbon atoms in the pyrrole rings at 397.7 eV.<sup>37</sup> Furthermore, the Cu 2p spectrum (Fig. 2c) reveals peaks at 954.46 eV and 934.6 eV, assignable to the Cu 2p<sub>1/2</sub> and Cu 2p<sub>3/2</sub> electronic states, respectively. Additionally, a characteristic satellite peak is observed in the Cu 2p region, providing further insight into the electronic structure of the copper atoms within CuPc.<sup>38</sup> There is no significant change in XPS spectra between both polymorphs of CuPc (Fig. S2 and S4†). The FT-IR spectrum

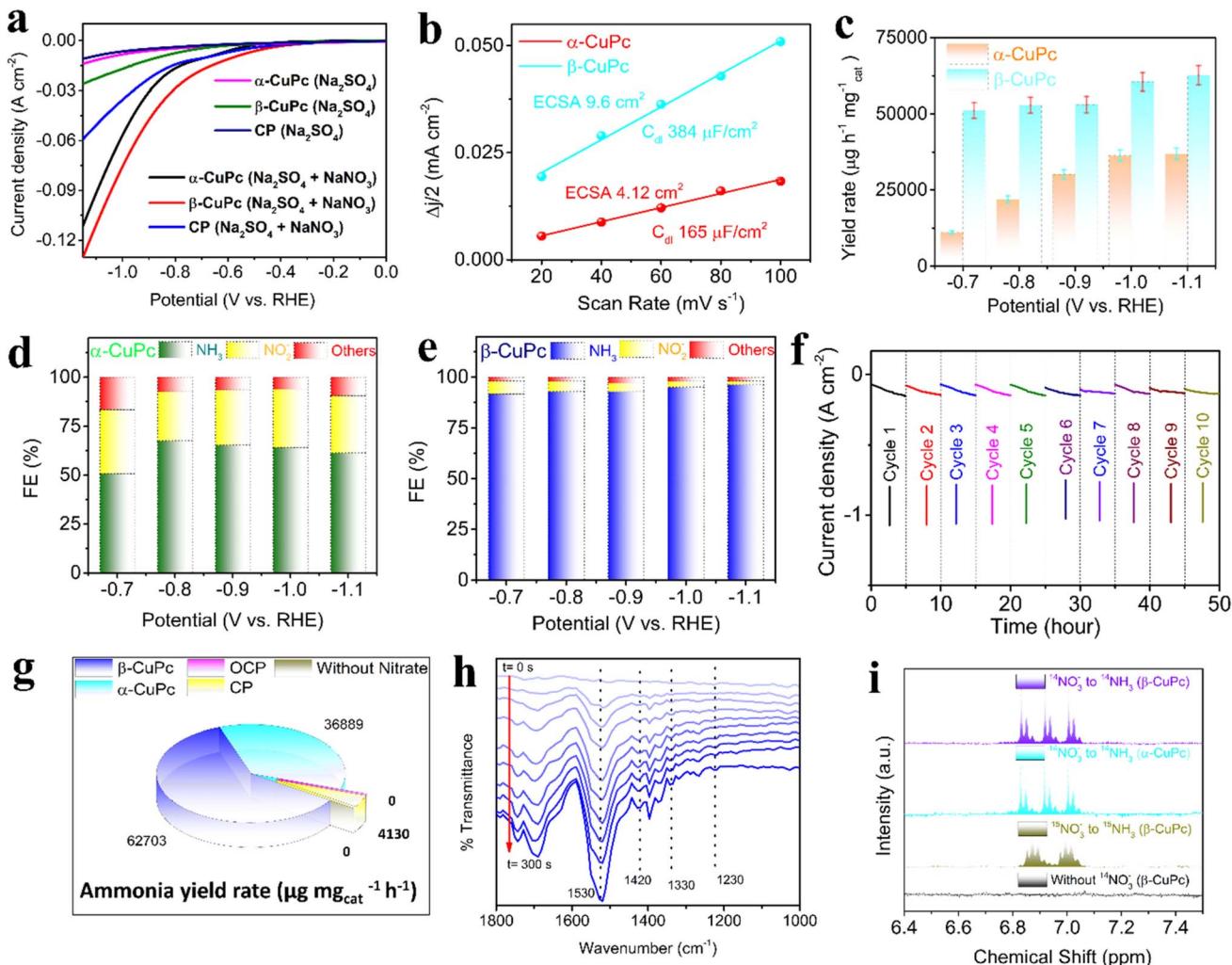
in Fig. 2d reveals distinct peaks corresponding to the vibrational characteristic modes. The C=N-C bonds at bridge sites showed characteristic peaks at  $1288$  cm<sup>-1</sup> with an additional peak at  $1334$  cm<sup>-1</sup> in CuPc.<sup>39</sup> The sharp peak around  $727$  cm<sup>-1</sup> corresponds to beta and the peak at  $720$  cm<sup>-1</sup> corresponds to alpha polymorphs.<sup>40</sup> UV-vis spectroscopy of  $\alpha$ -CuPc showed the Soret band and Q band at  $382$  nm and  $630$  nm, respectively (Fig. S1b and c†), whereas for the  $\beta$  polymorph, the Soret band and Q band shifted to a longer wavelength of  $450$  nm and  $653$  nm, respectively, due to better interaction, as mentioned before.

At the energy-scanning EXAFS beamline (BL-09) of the Indus-2 synchrotron source, XANES and EXAFS observations of the  $\beta$ -CuPc sample were conducted both before and after electrolysis. Fig. 2e shows the typical XANES spectra of  $\beta$ -CuPc recorded at the Cu *K*-edge, along with the metal Cu foil and CuO standard spectra. At about  $8985.5$  eV in the Cu *K*-edge XANES spectra, the CuPc catalyst exhibits a distinct Cu<sup>2+</sup> peak for the 1s-3d transition. As shown in Fig. 2f, the radial distribution function of the sample at the Cu *K*-edge is analyzed using the Fourier transform extended X-ray absorption fine structure (FT-EXAFS) method. The results show that the peak position at approximately  $\sim 1.5$  Å comes from the Cu-N bond, which corresponds to four neighboring N shells; the second peak, at  $\sim 2.5$  Å, has a contribution from a single scattered C shell with CN 8. The best-fit values of the samples are shown in Table S1† and the best-fit graphs (fitting range =  $0$ - $4$  Å) are shown in Fig. 2f. To determine the Cu-N and Cu-Cu bonds in CuPc and Cu foils, wavelet-transform extended X-ray absorption fine structure (WT-EXAFS) analysis was also performed (Fig. 2g-i). The WT-EXAFS study provides a convincing illustration of the above results.

### Electrochemical nitrate reduction

We performed all the experiments in an H-type cell with 3 electrode assemblies. Catalyst coated ( $0.145$  mg cm<sup>-2</sup>) carbon paper ( $1 \times 1$  cm<sup>2</sup>) was used as the working electrode, whereas for the counter and reference electrode, Pt foil and a saturated Ag/AgCl electrode were used respectively.  $0.25$  M NaNO<sub>3</sub> and  $0.1$  Na<sub>2</sub>SO<sub>4</sub> solutions were used as electrolytes. All potentials (Ag/AgCl) were converted to RHE using the equation  $E(\text{RHE}) = E(\text{Ag/AgCl}) + 0.198 + 0.059 \text{ pH}$ . Electrocatalytic nitrate reduction capability was first tested with linear sweep voltammetry at a scan rate of  $20$  mV s<sup>-1</sup> in a specially customised H-type cell for carbon paper (CP),  $\alpha$ -CuPc, and  $\beta$ -CuPc with  $0.1$  M Na<sub>2</sub>SO<sub>4</sub> as the supporting electrolyte in the presence and absence of  $0.25$  M NaNO<sub>3</sub> salt. In the case of carbon paper,  $\alpha$ -CuPc and  $\beta$ -CuPc current density increased at the same potential in the presence of NaNO<sub>3</sub> solution, indicating the reduction of nitrate, as found in Fig. 3a. The trend of increase in current density from carbon paper to  $\alpha$ -CuPc to  $\beta$ -CuPc clearly confirms the increase in electro-catalytical nitrate reduction activity. CV at different scan rates was performed to obtain the double layer capacitance of the catalysts (Fig. S7c and d†). The double layer capacitance of  $\beta$ -CuPc at  $0.5$  V vs. RHE was found to be twice that of  $\alpha$ -CuPc. Hence,  $\beta$ -CuPc had a higher electrochemical surface area than





**Fig. 3** (a) LSV of  $\alpha$ -CuPc,  $\beta$ -CuPc and carbon paper with and without nitrate. (b) Double layer capacitance of  $\alpha$ -CuPc and  $\beta$ -CuPc at 0.5 V vs. RHE at different scan rates. (c) Ammonia yield rate of  $\alpha$ -CuPc and  $\beta$ -CuPc at different potentials. (d and e) Faradaic efficiency of  $\alpha$ -CuPc and  $\beta$ -CuPc at different potentials. (f) Yield rate and FE for 10 cycles of 5 h for  $\beta$ -CuPc. (g) Yield percentage of ammonia of  $\alpha$ -CuPc,  $\beta$ -CuPc, and carbon paper at -1.1 V vs. RHE in the presence of 0.25 M  $\text{NaNO}_3$ , and ammonia yield percentage of  $\beta$ -CuPc at open-circuit potential and without nitrate solution. (h) *In situ* ATR-FTIR spectra during the nitrate reduction reaction at a constant potential. (i)  $^1\text{H}$  NMR spectra of both  $^{14}\text{NH}_4^+$  and  $^{15}\text{NH}_4^+$  produced from the nitrate reduction -1.1 V vs. RHE using  $^{14}\text{NO}_3^-$  and  $^{15}\text{NO}_3^-$ .

$\alpha$ -CuPC (Fig. 3b). The AC EIS study was performed within the frequency range of 1 to  $10^5$  Hz, and by fitting the spectroscopy data into Randall's circuit, we obtained two depressed, semicircle-like curves (Fig. S13†). From these curves, it is expected that the charge transfer resistance is much higher for  $\alpha$ -CuPc than  $\beta$ -CuPc. Product forming efficiency at different potentials was determined by chronoamperometry experiments (Fig. S7a and b†) for 1 h using the above-mentioned catalysts in the presence of nitrate and the probable nitrogenous products formed, *i.e.*, ammonia, nitrite, and hydrazine, were addressed by spectroscopic methods, described in the ESI.† The major product ammonia was measured by the indophenol blue method at 655 nm using UV-vis spectroscopy (Fig. S6†). We found that  $\beta$ -CuPc shows a higher yield and Faradaic efficiency than  $\alpha$ -CuPc for ammonia formation in all potentials.  $\beta$ -CuPc showed an ammonia yield rate of 62 703  $\mu\text{g h}^{-1} \text{mg}_{\text{cat}}^{-1}$  and

96% Faradaic efficiency, whereas  $\alpha$ -CuPc showed a yield rate of 36 889  $\mu\text{g h}^{-1} \text{mg}_{\text{cat}}^{-1}$  and 61% Faradaic efficiency (Fig. 3c) in 0.25 M  $\text{NaNO}_3$  and 0.1 M  $\text{Na}_2\text{SO}_4$  solutions, which is sufficiently higher than those of most of the catalysts mentioned in Table S3†. We checked for the ammonia formation rate and FE for the superior  $\beta$ -CuPc catalyst under low to high concentrations of nitrate salt and the experiment revealed that with an increase in nitrate concentration, the ammonia formation rate and FE increase, as shown in Fig. S15a.† We also performed experiments under different pH conditions. We found that for a fixed potential, the FE and yield rate increase with an increase in pH (0.05 M  $\text{H}_2\text{SO}_4$  to 0.1 M  $\text{Na}_2\text{SO}_4$  to 0.1 M  $\text{NaOH}$ ) (Fig. S15b†). In an acidic medium, the competitive HER is most prominent due to the presence of  $\text{H}^+$  ions in the medium, whereas with an increase in the pH value, this problem gets resolved. The presence of nitrite, the most prominent by-product, was confirmed

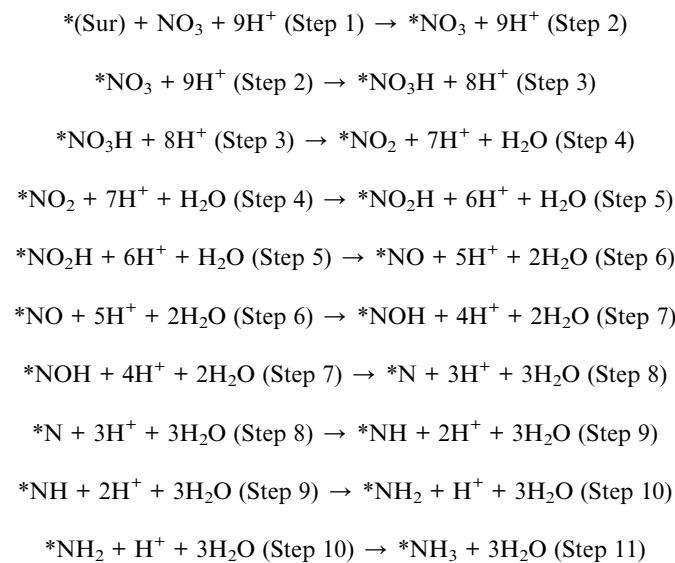
and quantified by spectroscopic analysis at  $\sim 550$  nm with the help of a complexing reagent *N*-(1-naphthyl)-ethylenediamine dihydrochloride (Fig. S10†). In the case of  $\alpha$ -CuPc, nitrite resulted in a maximum of 33% of FE, whereas in the case of  $\beta$ -CuPc, the value was around 6% of total FE at  $-0.7$  V vs. RHE (Fig. 3d and e). For  $\alpha$ -CuPc, with the increase in potential, the nitrite yield rate increased but for the  $\beta$ -CuPc catalyst, the nitrite yield rate decreased, which indicates that nitrite readily converted to the end product, ammonia, for  $\beta$ -CuPc as the ammonia yield also increased. The highest nitrite yield rate was found to be  $148\,278 \mu\text{g h}^{-1} \text{mg}_{\text{cat}}^{-1}$  for  $\alpha$ -CuPc at  $-1.1$  V, whereas for  $\beta$ -CuPc the highest rate was  $29\,934 \mu\text{g h}^{-1} \text{mg}_{\text{cat}}^{-1}$  at  $-0.7$  V (Fig. S14a†). Another nitrogenous by-product, hydrazine, was tested by UV-vis spectroscopy using the Watt and Chrissp method (ESI†). Only a few  $\mu\text{g h}^{-1} \text{mg}_{\text{cat}}^{-1}$  of hydrazine was found in both cases, which is almost 0.1% of the total FE of the catalysts (Fig. S14b†). We have also measured hydrogen evolution during the  $\text{NO}_3\text{RR}$  experiment by gas chromatography. Experiments revealed that with an increase in potential, the amount of evolved hydrogen increased for both  $\alpha$ -CuPc and  $\beta$ -CuPc, but  $\alpha$ -CuPc produced more hydrogen than  $\beta$ -CuPc at any potential, which again proved the selectivity of  $\beta$ -CuPc towards the nitrate to ammonia formation reaction (Fig. S16†). However, both catalysts contribute very little to their FE (less than 2%) for the HER up to  $-1.1$  V vs. RHE. The recyclability of the superior catalyst  $\beta$ -CuPc was checked through 10 cycles of 5 hour chronoamperometry with 0.25 M  $\text{NaNO}_3$  and 0.1 M  $\text{Na}_2\text{SO}_4$  at  $-1.1$  V vs. RHE in an H-type cell with 3 electrode assemblies. All cycles showed almost the same yield rate and Faradaic efficiency with no significant change in current density (Fig. 3f and S12†). The electrochemically used  $\beta$ -CuPc was characterized with the XPS technique and XANES, and no such change in their spectra was found in comparison with the pristine one (Fig. S3 and S5†). These results not only indicate its excellent stability but also demonstrate its very good recyclability. To assess its stability, we performed further nitrate reduction reactions continuously for 60 hours in the presence of 0.25 M  $\text{NaNO}_3$  and 0.1 M  $\text{Na}_2\text{SO}_4$  solution at  $-1.1$  V vs. RHE (Fig. S17†), where we found no significant change in current density, which itself supports the stability of the catalyst. Various control experiments were performed to confirm ammonia formation. Chronoamperometry at  $-1.1$  V vs. RHE with only  $\text{Na}_2\text{SO}_4$  solution and an OCP experiment with  $\text{NaNO}_3$  were performed, but both cases failed to produce ammonia (Fig. 3g and S11b†). Experiments with carbon paper as a catalyst were performed, but it did not produce significant ammonia as compared to  $\alpha$ -CuPc and  $\beta$ -CuPc. Fig. 3g shows the relative yield rate percentage of the respective conditions.

To identify the reaction intermediate, an *in situ* ATR-FTIR experiment was performed (Fig. 3h). The peaks at  $1330 \text{ cm}^{-1}$  and  $1420 \text{ cm}^{-1}$  were found to increase gradually with time and are assigned to hydrogenated  $-\text{NH}_2$  and  $\text{H}-\text{N}-\text{H}$ , respectively.<sup>41</sup> The peaks for deoxygenated intermediates like  $^*\text{NO}$  and  $^*\text{NO}_2$  emerge at  $1530 \text{ cm}^{-1}$  and  $1230 \text{ cm}^{-1}$ , respectively, with increasing time at a constant potential, confirming the reduction of  $\text{NO}_3^-$  to  $\text{NH}_3$  with various reaction intermediates.<sup>42</sup> After the electrolysis, whether the product is ammonia or not is

confirmed by the  $^1\text{H-NMR}$  experiment with  $^{14}\text{NO}_3^-$  and  $^{15}\text{NO}_3^-$  (Fig. 3i). Three peaks at 6.8, 6.9, and 7.0 ppm for  $\text{Na}^{14}\text{NO}_3$  and two peaks near 7 ppm for  $\text{Na}^{15}\text{NO}_3$  confirmed the formation of  $^{14}\text{NH}_3$  and  $^{15}\text{NH}_3$ , respectively, using catalysts. No peak was found when electrolysis was carried out without nitrate containing electrolyte (Fig. 3i). The ammonia formation rate was rechecked using ion chromatography (ESI†). UV-vis spectroscopy and IC techniques showed similar results, as shown in Fig. S11a.† The extraordinary catalytic properties and the selectivity of the reduced end product are due to the high electron transport properties of  $\beta$ -CuPc. The mentioned transport properties are due to the interaction of the d-orbital of Cu with the p-orbital of nitrogen of the above and below molecules, which are exactly situated above and below the central Cu atom in the case of  $\beta$ -CuPc, which is not possible in the case of the  $\alpha$ -polymorph.

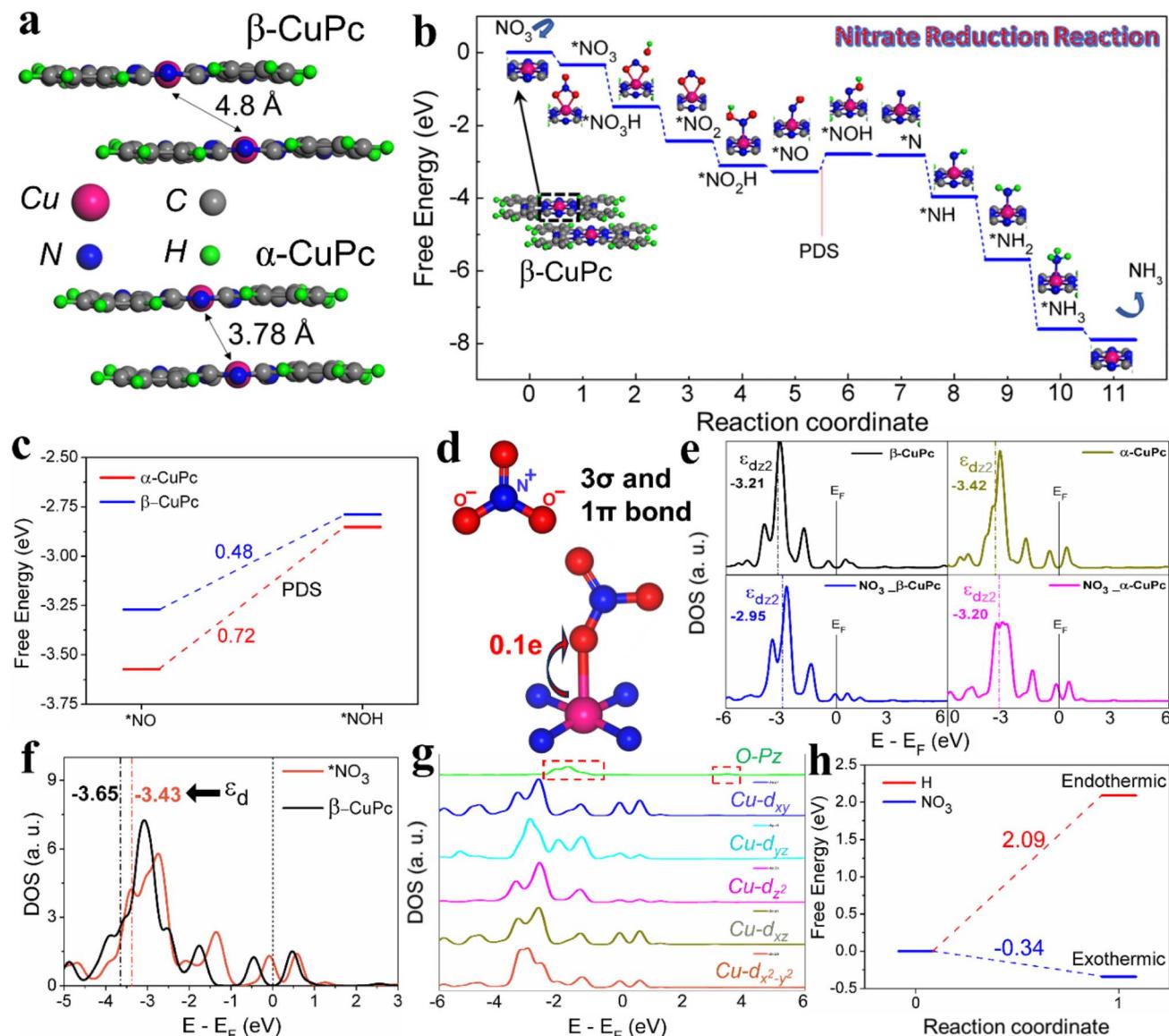
### Theoretical insight into the $\text{NO}_3\text{RR}$

DFT calculations are employed to investigate the catalytic performance of copper phthalocyanine (CuPc) based catalysts towards the nitrate reduction reaction ( $\text{NO}_3\text{RR}$ ). Two model structures,  $\alpha$ -CuPc and  $\beta$ -CuPc, are considered, as depicted in Fig. 4a and S18†.<sup>43</sup> The analysis revealed that the Cu site serves as the active site for capturing  $\text{NO}_3$  molecules, while the N, C, and H sites lack the capability to adsorb  $\text{NO}_3$  molecules. Thus, the Cu site of  $\alpha$ -CuPc and  $\beta$ -CuPc models is explored for comparing their  $\text{NO}_3\text{RR}$  performance. The  $\text{NO}_3\text{RR}$  mechanism involves the hydrogenation process of adsorbed  $\text{NO}_3$  molecules, leading to the production of ammonia ( $\text{NH}_3$ ) and water ( $\text{H}_2\text{O}$ ) molecules, as illustrated in Fig. S19†.<sup>7,8,44,45</sup> The reaction steps are given below.



Further, a DFT study is carried out to identify the reaction intermediates and estimate their total energies. The full free energy profile of the  $\text{NO}_3\text{RR}$  revealed that the  $^*\text{NO}$  to  $^*\text{NOH}$  conversion is the potential determining step (PDS) in both  $\alpha$ -CuPc and  $\beta$ -CuPc, as shown in Fig. 4b and c. The overpotential





**Fig. 4** (a) Model structures of  $\beta$ -CuPc and  $\alpha$ -CuPc. (b) The free energy profile of the NO<sub>3</sub>RR on the  $\beta$ -CuPc model. (c) The comparison of PDS. (d) The bonding mechanism in the NO<sub>3</sub> molecule and charge transfer between the Cu site of  $\beta$ -CuPc and the O atom of adsorbed NO<sub>3</sub>. (e) The projected density of states for  $\beta$ -CuPc and  $\alpha$ -CuPc with and without NO<sub>3</sub> adsorbed. E<sub>F</sub> is defined as the Fermi level at zero on the x-axis. (f) The projected density of states and d band center for the Cu active site in  $\beta$ -CuPc and NO<sub>3</sub> adsorbed  $\beta$ -CuPc. (g) The projected density of states for  $d_{xy}$ ,  $d_{yz}$ ,  $d_{zx}$ ,  $d_{z^2}$ , and  $d_{x^2-y^2}$  orbitals of the Cu active site and the p<sub>z</sub> orbital of the O atom in NO<sub>3</sub> adsorbed  $\beta$ -CuPc. (h) The comparison of NO<sub>3</sub> and hydrogen adsorption on the Cu active site of  $\beta$ -CuPc.

( $\eta$ ) values on the Cu site of  $\alpha$ -CuPc and  $\beta$ -CuPc are 0.72 eV and 0.48 eV, respectively, which mainly depend on the adsorption of the \*NO intermediate (Fig. 4c). Therefore,  $\beta$ -CuPc exhibits superior catalytic performance for the NO<sub>3</sub>RR due to the lower  $\eta$  value. The additional electronic structure analyses such as density of states, Bader charge analysis and charge density difference studies are performed to unveil the origin of their catalytic performance. The adsorption of the NO<sub>3</sub> molecule on the Cu active site is an initial step, essential for initiating the NO<sub>3</sub>RR mechanism. It is known that the NO<sub>3</sub> molecule consists of three  $\sigma$  bonds and one  $\pi$  bond between nitrogen and oxygen, as illustrated in Fig. 4d. During adsorption, the bonding

between the Cu site and the oxygen atom tends to transfer charge into the anti-bonding orbital ( $\pi^*$ ) of NO<sub>3</sub>, which makes the N-O bond weaker. Specifically, the Cu active site donates its d-electrons to the vacant  $\pi^*$  orbital of NO<sub>3</sub><sup>-</sup>, while simultaneously accepting electrons from NO<sub>3</sub><sup>-</sup> through an electron acceptance-donation mechanism. Using Bader charge analysis, we investigated the charge transfer between the Cu active site and the NO<sub>3</sub> molecule during adsorption on  $\beta$ -CuPc. The analysis revealed that a total of 0.1 e is transferred from the Cu active site to the NO<sub>3</sub> molecule (Fig. 4d). Also, a similar charge transfer is observed in the case of the Cu site of  $\alpha$ -CuPc. Additionally, density of states calculations are conducted to establish

a correlation with catalytic activity. In this analysis, the d-band center ( $\varepsilon_d$ ) is a well-known parameter used to define the catalytic activity of metal and metal oxide-based catalysts.<sup>46,47</sup> However, the  $\varepsilon_d$  is not particularly effective in explaining the catalytic performance of single-atom catalysts (SACs).<sup>48,49</sup> Also, we obtained a similar  $\varepsilon_d$  value of  $-3.65$  eV for the Cu active site in both  $\beta$ -CuPc and  $\alpha$ -CuPc. On the other hand, the  $d_z^2$  band center ( $\varepsilon_{dz^2}$ ) of metal atoms in SACs has been previously reported as a key factor in determining the oxygen adsorption energy.<sup>50,51</sup> Consequently, we calculated the  $\varepsilon_{dz^2}$  for the Cu site in both  $\beta$ -CuPc and  $\alpha$ -CuPc, obtaining values of  $-3.21$  eV and  $-3.42$  eV, respectively (Fig. 4e). This suggests that a more negative  $\varepsilon_{dz^2}$  correlates with stronger intermediate adsorption. As a result, the stronger \*NO adsorption and higher NO<sub>3</sub>RR overpotential observed on  $\alpha$ -CuPc can be attributed to its more negative  $\varepsilon_{dz^2}$  value compared to  $\beta$ -CuPc. Furthermore, the shift of their  $\varepsilon_{dz^2}$  toward the Fermi level following NO<sub>3</sub> adsorption indicates charge transfer between NO<sub>3</sub> and the Cu active site, a behaviour also observed in the  $\varepsilon_d$  of the Cu site after NO<sub>3</sub> adsorption (Fig. 4f). Fig. 4g presents the projected density of states (PDOS) for the O-p<sub>z</sub> orbital and the five sub-orbitals of the Cu-d-orbital, illustrating the hybridization between oxygen and the Cu site in  $\beta$ -CuPc. Moreover, the charge density difference helps to visualize the effect of the underlying CuPc on the Cu active site of both  $\alpha$ -CuPc and  $\beta$ -CuPc models. In Fig. S20,<sup>†</sup>  $\beta$ -CuPc displayed non-uniform charge redistribution with a slight charge accumulation on the Cu active site. In contrast, uniform charge redistribution is observed for  $\alpha$ -CuPc, with no charge difference on the Cu active site.

However, the hydrogen evolution reaction (HER) competes with the NO<sub>3</sub>RR, resulting in the production of H<sub>2</sub> molecules and hindering NO<sub>3</sub>RR efficiency.<sup>52</sup> Therefore, the hydrogen adsorption energy (a measure of the HER) on the Cu site of the  $\beta$ -CuPc catalyst is computed. As shown in Fig. 4h, it is observed that hydrogen adsorption is an endothermic process with an energy of  $+2.09$  eV, while NO<sub>3</sub> adsorption energy is  $-0.34$  eV, indicating an exothermic and favourable process. Thus, it is demonstrated that the NO<sub>3</sub>RR is the dominant process on the  $\beta$ -CuPc catalyst compared to the HER. Additionally, the higher endothermic energy of hydrogen adsorption prevents active sites from experiencing hydrogen poisoning.

## Conclusion

In conclusion, sandwiched structured  $\beta$ -CuPc showed an ammonia yield rate of  $62\,703\, \mu\text{g h}^{-1}\, \text{mg}_{\text{cat}}^{-1}$  and 96% Faradaic efficiency, whereas less well aligned  $\alpha$ -CuPc showed a yield rate of  $36\,889\, \mu\text{g h}^{-1}\, \text{mg}_{\text{cat}}^{-1}$  and 61% Faradaic efficiency at  $-1.1$  V vs. RHE under ambient conditions. An STM/S experiment demonstrates that the well-aligned stacked  $\beta$ -CuPc possesses enhanced transport capabilities attributed to excellent interactions of parallel molecules within its one-dimensional nanostructure, as evidenced by its electrocatalytic performance. Theoretical research implies the NO<sub>3</sub>RR is the predominant reaction on the  $\beta$ -CuPc catalyst in comparison to the HER, which is also verified experimentally. This study presents an efficient NO<sub>3</sub>RR process utilizing CuPc nanostructures with

different molecular alignments to produce ammonia at room temperature and may also pave the way for the development of cost-effective metal complexes.

## Data availability

The data are available within the article and its ESI.<sup>†</sup> The data that support the findings of this study are available on request from the corresponding author.

## Author contributions

SB synthesized the materials and carried out all the electrochemical measurements and analysed the results. AM performed the STM experiment and analyzed the results. SK and RT performed the DFT calculation. RT, SK, SB, and UKG analyzed the DFT results. SB and AA analyzed the NMR, XANES and XAFS results. SB wrote the original manuscript. AA, AM, SK, RT, SM and UKG co-wrote the manuscript. All authors approved the final version of the manuscript. UKG supervised the entire project.

## Conflicts of interest

The authors declare no competing interests.

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## References

- 1 L. Bai, F. Franco, J. Timoshenko, C. Rettenmaier, F. Scholten, H. S. Jeon, A. Yoon, M. Rüscher, A. Herzog, F. T. Haase, S. Kühl, S. W. Chee, A. Bergmann and R. C. Beatriz, *J. Am. Chem. Soc.*, 2024, **146**, 9665–9678.
- 2 G. Zhang, X. Li, K. Chen, Y. Guo, D. Ma and K. Chu, *Angew. Chem., Int. Ed.*, 2023, **62**, e202300054.
- 3 H. Zhang, H. Wang, X. Cao, M. Chen, Y. Liu, Y. Zhou, M. Huang, L. Xia, Y. Wang, T. Li, D. Zheng, Y. Luo, S. Sun, X. Zhao and X. Sun, *Adv. Mater.*, 2024, **36**, 2312746.
- 4 A. Adalder, S. Paul, N. Barman, A. Bera, S. Sarkar, N. Mukherjee, R. Thapa and U. K. Ghorai, *ACS Catal.*, 2023, **13**, 13516–13527.
- 5 Q. Liu, L. Xie, J. Liang, Y. Ren, Y. Wang, L. Zhang, L. Yue, T. Li, Y. Luo, N. Li, B. Tang, Y. Liu, S. Gao, A. A. Alshehri,



- I. Shakir, P. O. Agboola, Q. Kong, Q. Wang, D. Ma and X. Sun, *Small*, 2022, **18**, 2106961.
- 6 G. Zhang, G. Wang, Y. Wan, X. Liu and K. Chu, *ACS Nano*, 2023, **17**, 21328–21336.
- 7 S. Paul, S. Sarkar, A. Adalder, S. Kapse, R. Thapa and U. K. Ghorai, *ACS Sustain. Chem. Eng.*, 2023, **11**, 6191–6200.
- 8 S. Sarkar, A. Adalder, S. Paul, S. Kapse, R. Thapa and U. K. Ghorai, *Appl. Catal., B*, 2024, **343**, 123580.
- 9 B. Zhang, Z. Dai, Y. Chen, M. Cheng, H. Zhang, P. Feng, B. Ke, Y. Zhang and G. Zhang, *Nat. Commun.*, 2024, **15**, 1–14.
- 10 R. Daiyan, T. Tran-Phu, P. Kumar, K. Iputera, Z. Tong, J. Leverett, M. H. A. Khan, A. Asghar Esmailpour, A. Jalili, M. Lim, A. Tricoli, R. S. Liu, X. Lu, E. Lovell and R. Amal, *Energy Environ. Sci.*, 2021, **14**, 3588–3598.
- 11 Q. Gao, H. S. Pillai, Y. Huang, S. Liu, Q. Mu, X. Han, Z. Yan, H. Zhou, Q. He, H. Xin and H. Zhu, *Nat. Commun.*, 2022, **13**, 1–12.
- 12 H. Luo, S. Li, Z. Wu, M. Jiang, M. Kuang, Y. Liu, W. Luo, D. Zhang and J. Yang, *Adv. Funct. Mater.*, 2024, **34**, 2403838.
- 13 Y. H. Huang and T. C. Zhang, *Water Res.*, 2005, **39**, 1751–1760.
- 14 X. Shi, M. Xie, K. Yang, Y. Niu, H. Ma, Y. Zhu, J. Li, T. Pan, X. Zhou, Y. Cui, Z. Li, Y. Yu, X. Yu, J. Ma and H. Cheng, *Angew. Chem., Int. Ed.*, 2024, **63**, e202406750.
- 15 Y. Wang, F. Hao, M. Sun, M. T. Liu, J. Zhou, Y. Xiong, C. Ye, X. Wang, F. Liu, J. Wang, P. Lu, Y. Ma, J. Yin, H. C. Chen, Q. Zhang, L. Gu, H. M. Chen, B. Huang and Z. Fan, *Adv. Mater.*, 2024, **36**, 2313548.
- 16 X. Yang, R. Ahuja and W. Luo, *Nano Energy*, 2023, **113**, 108557.
- 17 Y. Y. Lou, Q. Z. Zheng, S. Y. Zhou, J. Y. Fang, O. Akdim, X. Y. Ding, R. Oh, G. S. Park, X. Huang and S. G. Sun, *ACS Catal.*, 2024, **14**, 5098–5108.
- 18 R. Qi, L. Zhang, S. Ren, B. Shi, M. Zhong, Z. J. Chen and X. Lu, *Nano Lett.*, 2024, **24**, 8964–8972.
- 19 H. Li, N. Ma, Y. Long, X. Tang, W. Ou, F. Lyu, J. Liu, B. Zhou, J. Fan, J. Lu and Y. Y. Li, *ACS Appl. Mater. Interfaces*, 2024, **16**, 46312–46322.
- 20 Z. Wu, Y. Song, H. Guo, F. Xie, Y. Cong, M. Kuang and J. Yang, *Interdiscip. Mater.*, 2024, **3**, 245–269.
- 21 H. Zhang, C. Wang, H. Luo, J. Chen, M. Kuang and J. Yang, *Angew. Chem., Int. Ed.*, 2023, **62**, e202217071.
- 22 S. Keshipour and S. Mohammad-Alizadeh, *Sci. Rep.*, 2021, **11**, 1–8.
- 23 A. Adalder, S. Paul, B. Ghorai, S. Kapse, R. Thapa, A. Nagendra and U. K. Ghorai, *ACS Appl. Mater. Interfaces*, 2023, **15**, 34642–34650.
- 24 S. Paul, A. Adalder, N. Barman, R. Thapa, A. Bera, K. Mitra and U. K. Ghorai, *Adv. Funct. Mater.*, 2024, 2408314.
- 25 A. Adalder, K. Mitra, N. Barman, R. Thapa, S. Bhowmick and U. K. Ghorai, *Adv. Energy Mater.*, 2024, 2403295.
- 26 J. Mukherjee, S. Paul, A. Adalder, S. Kapse, R. Thapa, S. Mandal, B. Ghorai, S. Sarkar and U. K. Ghorai, *Adv. Funct. Mater.*, 2022, **32**, 2200882.
- 27 U. K. Ghorai, S. Paul, B. Ghorai, A. Adalder, S. Kapse, R. Thapa, A. Nagendra and A. Gain, *ACS Nano*, 2021, **15**, 5230–5239.
- 28 J. Mukherjee, A. Adalder, N. Mukherjee and U. K. Ghorai, *Catal. Today*, 2023, **423**, 113905.
- 29 N. Mukherjee, A. Adalder, N. Barman, R. Thapa, R. Urkude, B. Ghosh and U. K. Ghorai, *J. Mater. Chem. A*, 2024, **12**, 3352–3361.
- 30 S. Bhardwaj, S. K. Das, A. Biswas, S. Kapse, R. Thapa and R. S. Dey, *Chem. Sci.*, 2023, **14**, 8936–8945.
- 31 U. K. Ghorai, S. Das, S. Saha, N. Mazumder, D. Sen and K. K. Chattopadhyay, *Dalton Trans.*, 2014, **43**, 9260–9266.
- 32 K. P. Madhuri, P. Kaur, M. E. Ali and N. S. John, *J. Phys. Chem. C*, 2017, **121**, 9249–9259.
- 33 I. Arbi, B. Ben Hamada, A. Souissi, S. Menzli, C. Ben Azzouz, A. Laribi, A. Akremi and C. Chefí, *Appl. Surf. Sci.*, 2014, **305**, 396–401.
- 34 K. Hunger, *Rev. Prog. Color. Relat. Top.*, 1999, **29**, 71–84.
- 35 S. Dutt, A. R. Kottaichamy, N. C. Dargily, S. Mukhopadhyay, B. Nayak, M. C. Devendrachari, C. P. Vinod, H. M. Nimbegondi Kotresh and M. Ottakam Thotiyil, *Chem. Sci.*, 2024, **15**, 13262–13270.
- 36 J. M. Assour and W. K. Kahn, *J. Am. Chem. Soc.*, 1965, **87**, 207–212.
- 37 B. Das, M. Samanta, P. Sarkar, U. K. Ghorai, A. Mallik and K. K. Chattopadhyay, *Adv. Electron. Mater.*, 2021, **7**, 2001079.
- 38 C. Jiang, H. Xue, T. Wang and J. He, *ChemCatChem*, 2023, **15**, e202201631.
- 39 J. S. Jung, J. W. Lee, K. Kim, M. Y. Cho, S. G. Jo and J. Joo, *Chem. Mater.*, 2010, **22**, 2219–2225.
- 40 A. K. Hassan and R. D. Gould, *Phys. Status Solidi A*, 1992, **132**, 91–101.
- 41 X. Li, P. Shen, X. Li, D. Ma and K. Chu, *ACS Nano*, 2023, **17**, 1081–1090.
- 42 N. Zhang, G. Zhang, P. Shen, H. Zhang, D. Ma and K. Chu, *Adv. Funct. Mater.*, 2023, **33**, 2211537.
- 43 T. Zou, X. Wang, H. Ju, L. Zhao, T. Guo, W. Wu and H. Wang, *Crystals*, 2018, **8**, 22.
- 44 R. Zhang, C. Li, H. Cui, Y. Wang, S. Zhang, P. Li, Y. Hou, Y. Guo, G. Liang, Z. Huang, C. Peng and C. Zhi, *Nat. Commun.*, 2023, **14**, 1–11.
- 45 N. C. Kani, J. A. Gauthier, A. Prajapati, J. Edgington, I. Bordawekar, W. Shields, M. Shields, L. C. Seitz, A. R. Singh and M. R. Singh, *Energy Environ. Sci.*, 2021, **14**, 6349–6359.
- 46 B. Hammer and J. K. Norskov, *Nature*, 1995, **376**(6537), 238–240.
- 47 B. Hammer and J. K. Nørskov, *Adv. Catal.*, 2000, **45**, 71–129.
- 48 V. Fung, G. Hu, Z. Wu and D. E. Jiang, *J. Phys. Chem. C*, 2020, **124**, 19571–19578.
- 49 G. Di Liberto, L. A. Cipriano and G. Pacchioni, *ACS Catal.*, 2022, **12**, 5846–5856.
- 50 Y. Wang, Y. Liang, T. Bo, S. Meng and M. Liu, *J. Phys. Chem. Lett.*, 2022, **13**, 5969–5976.
- 51 Z. Fu, B. Yang and R. Wu, *Phys. Rev. Lett.*, 2020, **125**, 156001.
- 52 Z. Shu, H. Chen, X. Liu, H. Jia, H. Yan and Y. Cai, *Adv. Funct. Mater.*, 2023, **33**, 2301493.

