

Showcasing research from Professor Binghui Xu's laboratory, Institute of Materials for Energy and Environment, College of Materials Science and Engineering, Qingdao University 266071, China.

Synthesis of a MOF-derived magnetite quantum dots on surface modulated reduced graphene oxide composite for high-rate lithium-ion storage

Reactions between 1,3,5-benzenetricarboxylic acid ( $H_3BTC$ ), tea polyphenol (TP), graphene oxide (GO) and iron foils are triggered under mild aqueous conditions. Both  $H_3BTC$  and TP are decorated on the surface of GO sheets by  $\pi$ - $\pi$  conjugation while the redox reaction between Fe and GO takes place. The coordination between  $Fe^{3+}$  and  $BTC^{3+}$  leads to *in situ* crystallization of Fe-BTC metal-organic framework (MOF) on the surface of the TP-decorated reduced graphene oxide (TP-RGO), both the size of Fe-BTC MOF domains and the restacking of TP-RGO sheets are significantly controlled.

### As featured in:



See Binghui Xu *et al.*,  
RSC Appl. Interfaces, 2024, 1, 233.



Cite this: *RSC Appl. Interfaces*, 2024, 1, 233

## Synthesis of a MOF-derived magnetite quantum dots on surface modulated reduced graphene oxide composite for high-rate lithium-ion storage

Ruixin Jia, Longbiao Yu, Zhenqi Han, Shuo Liu, Panpan Shang, Siqi Deng, Xuehua Liu and Binghui Xu \*

As an anodic candidate material for lithium-ion batteries (LIBs), magnetite ( $\text{Fe}_3\text{O}_4$ ) has the advantages of high theoretical capacity ( $926 \text{ mA h g}^{-1}$ ), environmental friendliness, and natural abundance. The critical problems of inferior rate capability and cycling stability for  $\text{Fe}_3\text{O}_4$  anodes need to be urgently solved. In this work, a composite anode material with  $\text{Fe}_3\text{O}_4$  quantum dots immobilized by pyrolytic carbon and reduced graphene oxide skeleton ( $\text{Fe}_3\text{O}_4$  QDs@C/RGO) is rationally engineered from a metal-organic framework (MOF) domain on surface-modulated RGO precursor. The involved raw materials are 1,3,5-benzenetricarboxylic acid ( $\text{C}_9\text{H}_6\text{O}_6$ , trimesic acid,  $\text{H}_3\text{BTC}$ ), tea polyphenol (TP), few-layered graphene oxide (GO) and metal iron (Fe) foils, which react in mild hydrothermal condition in the synthesis of the Fe-BTC/TP-RGO precursor sample. The essence of the reactions has been uncovered, and the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite shows excellent lithium-ion storage capability with a reversible capacity of  $1070.84 \text{ mA h g}^{-1}$  after 150 cycles at  $200 \text{ mA g}^{-1}$ ,  $738.03 \text{ mA h g}^{-1}$  after 300 cycles at  $1000 \text{ mA g}^{-1}$  in half cells, and  $717.29 \text{ mA h g}^{-1}$  after 150 cycles at  $200 \text{ mA g}^{-1}$  in full cells. The sample synthesis protocol has the merits of simplicity and eco-friendliness, and the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample can be regarded as a promising anodic candidate for high-performance LIBs.

Received 6th August 2023,  
Accepted 3rd October 2023

DOI: 10.1039/d3lf00128h

rsc.li/RSCApplInter

## 1. Introduction

Lithium-ion batteries (LIBs) currently have a dominant position in the portable electronics and new-energy automobile markets primarily from their higher energy density and superior cycling performance.<sup>1–3</sup> To overcome the obstacle of the theoretical capacity limit of graphitic anodes ( $372 \text{ mA h g}^{-1}$ ), both academia and industry are striving to develop advanced anode materials to satisfy the increasing demand from the rapid development of renewable energy utilization.<sup>4–12</sup> Magnetite ( $\text{Fe}_3\text{O}_4$ ) has been extensively studied as an alternative anode material for LIBs due to its high theoretical capacity ( $926 \text{ mA h g}^{-1}$ ), environmental friendliness, and natural abundance. However, the sluggish electron conductivity and heavy volume change always lead to inferior cycling stability, particularly under higher current rate.<sup>13,14</sup> In recent studies, quantum dots (QDs) with ultra-small size as the electrochemically active materials in LIBs have been proven to be able to shorten the ion diffusion distance, reduce the internal strain, and improve the reaction

conversion efficiency.<sup>15–18</sup> Meanwhile, to achieve satisfactory lithium-ion storage performances at high current densities, effectively immobilizing  $\text{Fe}_3\text{O}_4$  QDs with a unique conductive supporting framework is a challenging task.

Two-dimensional graphene oxide (GO) with a large specific surface area and abundant functional groups is dispersible in water phase, which facilitates the synthesis of reduced graphene oxide (RGO)-supported metal oxide nanocomposite materials.<sup>19,20</sup> However, the excessive size growth of the nanocrystals and over-restacking of the RGO sheets impedes to a large extent the engineering of  $\text{Fe}_3\text{O}_4$  QDs on RGO with optimized microstructure. On the other hand, porous metal-organic framework (MOF) materials are usually constituted of periodically arranged metal ion centers and organic ligands from a coordinating chelation reaction, which are also being employed as precursors for the synthesis of pyrolytic carbon-supported metal oxide nanocomposites.<sup>21–25</sup> In our previous work, 1,3,5-benzenetricarboxylic acid ( $\text{C}_9\text{H}_6\text{O}_6$ ,  $\text{H}_3\text{BTC}$ ), a water-soluble small-molecule organic ligand, has been introduced in a redox coprecipitation reaction between metallic Fe and GO sheets to synthesize Fe-BTC MOF domains on a RGO sheets precursor sample, which is finally converted to  $\text{Fe}_3\text{O}_4$  nanoparticles on carbon matrix after thermal treatment.<sup>26</sup> Therefore, it is of great significance to further explore a simplified and eco-friendly method to synthesize a

Institute of Materials for Energy and Environment, College of Materials Science and Engineering, Qingdao University, Qingdao 266071, China.  
E-mail: xubinghuiqdu@qdu.edu.cn, xubinghuisu@163.com



carbonaceous matrix-supported  $\text{Fe}_3\text{O}_4$  QDs composite on the above foundation.

Tea polyphenol (TP) is an extract from green tea and a water-soluble mixture of condensed tannins with main-chain molecules consisting mainly of rigid aromatic rings. TP is effective in modifying the surface properties of RGO sheets, which can act as a reducing agent for partial removal of the functional groups on GO and directly decorate on RGO sheets.<sup>27-29</sup> In this work, as illustrated in Scheme 1, under mild aqueous conditions,  $\text{H}_3\text{BTC}$ , TP, GO, and iron foils are directly used as starting materials. Both  $\text{H}_3\text{BTC}$  and TP molecules are decorated on the surface of the GO sheets *via* the  $\pi$ - $\pi$  conjugation effect while a redox reaction between  $\text{Fe}^{3+}$  and the  $\text{BTC}^{3+}$  leads to the *in situ* crystallization of the Fe-BTC MOF domains on the TP-decorated RGO (TP-RGO) framework. Particularly, in the presence of TP, both the crystallization of the Fe-BTC MOF domains and the restacking of the TP-RGO sheets are significantly controlled. In other words, the as-formed small-sized Fe-BTC domains and abundant TP molecules are jointly dispersed on RGO sheets without severe restacking in the Fe-BTC/TP-RGO precursor. After thermal treatment, the composite sample with  $\text{Fe}_3\text{O}_4$  QDs immobilized by pyrolytic carbon and RGO skeleton ( $\text{Fe}_3\text{O}_4$  QDs@C/RGO) is finally synthesized, in which the homogeneously distributed small-sized Fe-BTC domains are converted to pyrolytic carbon-immobilized  $\text{Fe}_3\text{O}_4$  QD clusters on RGO sheets. The unique microstructure of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample contributes to improved electronic and ionic transportation efficiency and suppressed

volume change; thus, significantly enhanced electrochemical performances for this sample have been witnessed.

## 2. Experimental section

### 2.1 Reagents and materials

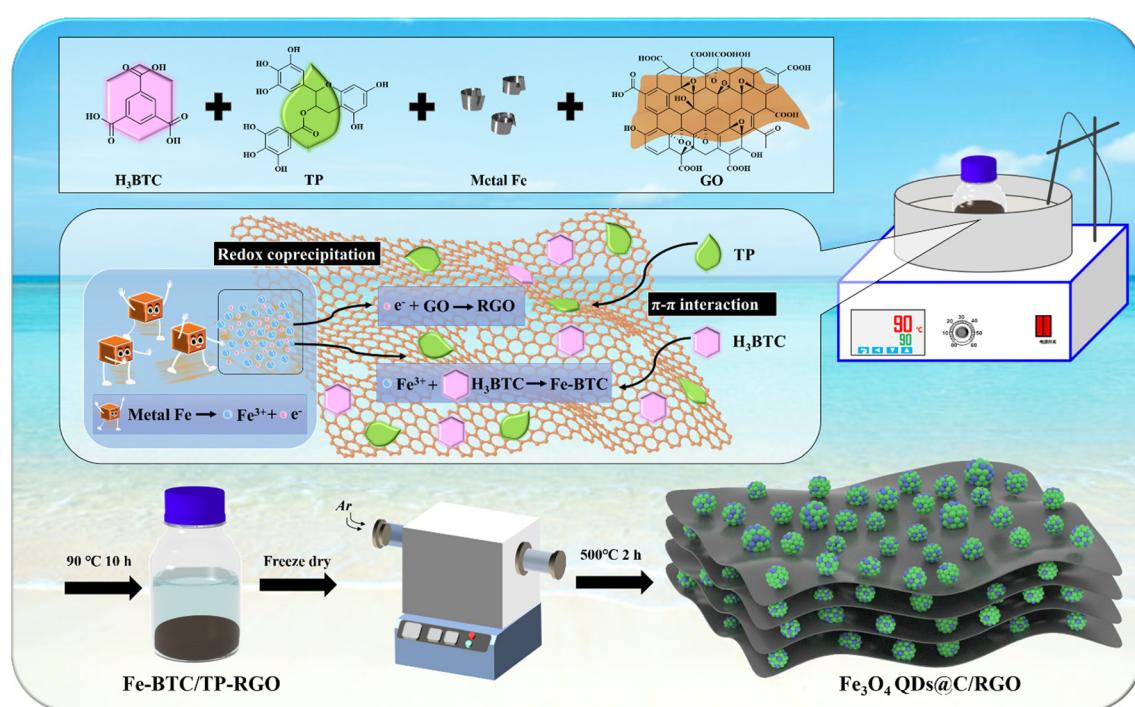
$\text{H}_3\text{BTC}$  (98%) was supplied by Accela Chembio. TP (97%) was obtained from Macklin. Metal Fe foil (0.15 mm thick, 99.99%) was purchased from Hunan Metal Materials, China. GO was synthesized from natural graphite flake (Sigma Aldrich, 325 mesh) according to our previously published method.<sup>30</sup>

### 2.2 Sample synthesis

$\text{Fe}_3\text{O}_4$  QDs@C/RGO: in a typical experiment,  $\text{H}_3\text{BTC}$  (1.0 g), TP (1.0 g), GO (0.1 g) and several polished iron foils were dissolved in 150 mL of deionized water and kept magnetically stirred in an oil bath at 90 °C for 10 h. After freeze-drying, the intermediate Fe-BTC/TP-RGO powder with black color was obtained. Finally, the Fe-BTC/TP-RGO sample underwent a thermal treatment at 500 °C for 2 h in Ar condition, and the final black  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample was thus fabricated.

### 2.3 Sample characterization and electrochemical measurement

**2.3.1 Sample characterization.** X-ray diffraction (XRD) patterns were measured with an X-ray diffractometer (Rigaku Ultima IV with  $\text{Cu K}\alpha$  radiation,  $\lambda = 0.15418$  nm). Fourier transform infrared spectroscopy (FTIR) was conducted in the



**Scheme 1** Schematic illustration for the synthesis of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample.



region 400–4000  $\text{cm}^{-1}$  using a Nicolet iS50 FTIR spectrophotometer. A Renishaw inVia Plus Micro-Raman spectroscopy system equipped with a 50 mW DPSS laser at 532 nm was used to record the Raman spectra of the samples. Thermogravimetric analyses (TGA) were performed with a Mettler Toledo TGA-2 in air with a heating rate of 10  $^{\circ}\text{C min}^{-1}$ . A PHI Quantera II spectrometer was used to carry out X-ray photoelectron spectroscopy (XPS). A Quantachrome Autosorb-IQ3 was used to measure the nitrogen adsorption/desorption isotherms, and the specific surface area was based on the Brunauer–Emmett–Teller (BET) method. Moreover, the corresponding pore size distributions were calculated based on the Barrett–Joyner–Halenda (BJH) model. Field-emission scanning electron microscopy (FESEM, JEOL Model JSM-7800F) and transmission electron microscopy (TEM, JEOL Model JEM-2100) were used to reveal the morphologies and microstructures.

**3.2.3 Electrochemical measurement.** For the assembly of coin-type half cells, a slurry was mixed with  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode materials, Super P, polyvinylidene fluoride (PVDF), and carbon fiber (CF) in a weight ratio of 7:1:1:1. An appropriate amount of 1-methyl-2-pyrrolidone (NMP) was added to form a homogeneous slurry, which was coated on copper foil by the doctor blade technique. For the  $\text{LiFePO}_4$  cathode,  $\text{LiFePO}_4$ , Super P and PVDF were mixed in a weight ratio of 8:1:1, then NMP was added to form a homogeneous slurry, which was coated on an aluminum foil. CR2016 half cells were assembled with metal lithium foil as counter/reference electrode. Celgard 2600 film was used as a separator between the two electrodes. The electrolyte was composed of 1.0 M lithium hexafluorophosphate ( $\text{LiPF}_6$ ) in a solvent of ethylene carbonate (EC), dimethyl carbonate (DMC), and ethylene methyl carbonate (EMC) in a volumetric ratio of 1:1:1, in which 1.0% vinylene carbonate (VC) was added. For the coin-type full cells, CR2025 cells were assembled using a  $\text{LiFePO}_4$  cathode and an  $\text{Fe}_3\text{O}_4$  QDs@C/

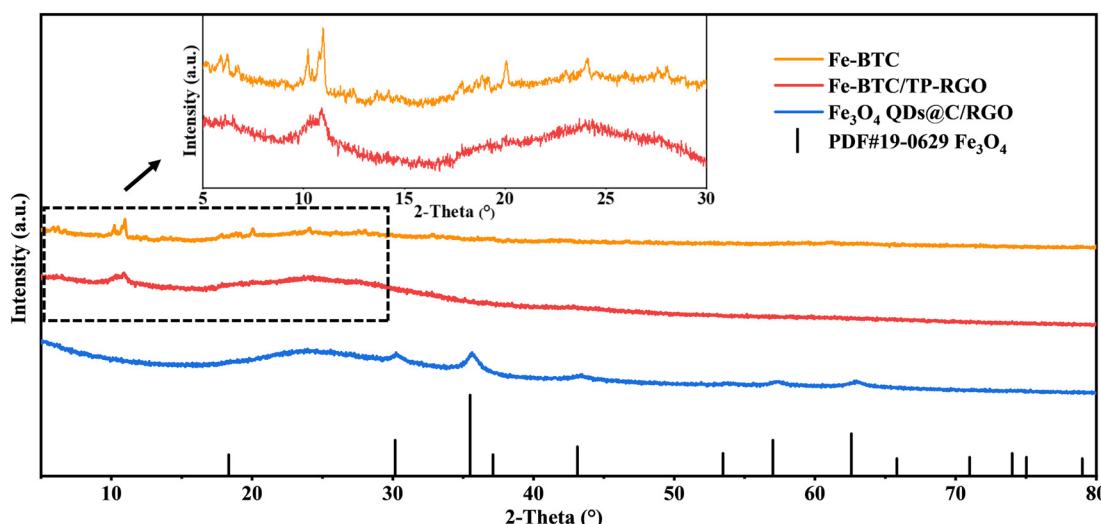
RGO anode, in which the anode was pre-lithiated for 6 cycles in a half cell.

Cyclic voltammetry (CV) measurements were performed with a CHI 660E electrochemical workstation. Galvanostatic charge and discharge tests were carried out with a Neware battery test instrument and the specific capacity was calculated based on the mass of the synthesized composite samples. Electrochemical impedance spectroscopy (EIS) tests were also carried out using the CHI 660E electrochemical workstation.

### 3. Results and discussion

#### 3.1 Structure and morphology

In the first mild aqueous condition, the color of the reaction system gradually turns black, and the Fe-BTC/TP-RGO sample can be collected as a sediment. After freeze-drying and thermal treatment, the final  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample can be synthesized from the Fe-BTC/TP-RGO precursor. From the XRD pattern of the Fe-BTC sample in Fig. 1, several weak diffraction peaks located at  $2\theta = 10.16$ , 10.90, 20.06, and 23.98 degrees are in general agreement with Fe-BTC MOF.<sup>31–33</sup> The XRD pattern of Fe-BTC/TP-RGO displays a broad diffraction peak at  $2\theta = 24.50$  degrees indicating the formation of the RGO supporting skeleton.<sup>34</sup> Moreover, from the locally magnified XRD patterns for the above two samples, the characterization peaks are in good agreement, indicating the successful engineering of the Fe-BTC MOF domains in the Fe-BTC/TP-RGO sample. However, the significantly weakened intensity of the XRD peaks for the Fe-BTC MOF species is highly attributed to its limited crystallization on the surface of the TP-modulated RGO sheets. For the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample, the apparent diffraction peaks situated at  $2\theta$  of 30.10, 35.42, 43.05, 53.39, 56.94, and 62.52 degrees correspond to the  $\text{Fe}_3\text{O}_4$  (JCPDS card number 19-0629) crystallographic planes of (2 2 0), (3 1 1),



**Fig. 1** XRD patterns of the Fe-BTC, Fe-BTC/TP-RGO and  $\text{Fe}_3\text{O}_4$  QDs@C/RGO samples.



(4 0 0), (4 2 2), (5 1 1), and (4 4 0), respectively. The characterization peak of RGO can also be seen without any diffraction peaks of other possible impurities.

The FTIR analysis result of the Fe-BTC/TP-RGO sample is shown in Fig. 2a. The sharp peak at about  $3388\text{ cm}^{-1}$  is attributed to free O-H bonds in the structure. The bands at  $2852\text{ cm}^{-1}$ ,  $1097\text{ cm}^{-1}$  and  $1636\text{ cm}^{-1}$  are attributed to the stretching vibrational peaks of C-H, C-O-C and C=C, while the peaks at  $1636$  and  $1386\text{ cm}^{-1}$  belong to aromatic C=C bonds and C-OH bonds, respectively.<sup>35,36</sup> Moreover, the characteristic peak at  $532\text{ cm}^{-1}$  is related to Fe-O stretching vibrations.

In Fig. 2b, two main Raman peaks of the D-band ( $1325\text{ cm}^{-1}$ ) and G-band ( $1590\text{ cm}^{-1}$ ) can be seen for the Fe-BTC/TP-RGO and  $\text{Fe}_3\text{O}_4$  QDs@C/RGO samples. The D-band corresponds to the defective and disordered part of the  $\text{sp}^3$  carbon, while the G-band is associated with the ordered  $\text{sp}^2$  carbon.<sup>37,38</sup> The integral intensity ratio of the D-band and G-band is another feature of the Raman spectrum, which provides useful information about the degree of carbon

disorder. The intensity ratios  $I_D$  and  $I_G$  are measured to be 0.94 and 0.64 for the two samples, indicating fewer defects and a high degree of carbon order after thermal treatment. It can be revealed that with the presence of the surface-decorated TP, the conjugated electronic structure of the RGO sheets is repaired during the thermal treatment. Moreover, the large amount of ligand BTC pyrolyzed carbon also accounts for the high graphitization degree of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample.

Fig. 2c exhibits the TGA and DTG curves of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample. From the TGA curve, the initial weight loss that occurred from  $30\text{ }^\circ\text{C}$  to  $100\text{ }^\circ\text{C}$  is ascribed to the evaporation of water. The subsequent weight loss mainly occurs between  $300\text{ }^\circ\text{C}$  and  $600\text{ }^\circ\text{C}$ , which can be attributed to the oxidation of  $\text{Fe}_3\text{O}_4$  to  $\text{Fe}_2\text{O}_3$  and the decomposition of pyrolytic carbon. The mass loss of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample can be divided into two stages, probably originating from the decomposition of the organic pyrolyzed carbon and RGO sheets. From the corresponding DTG curve, the inflection points of the two stages are measured to be 369.2

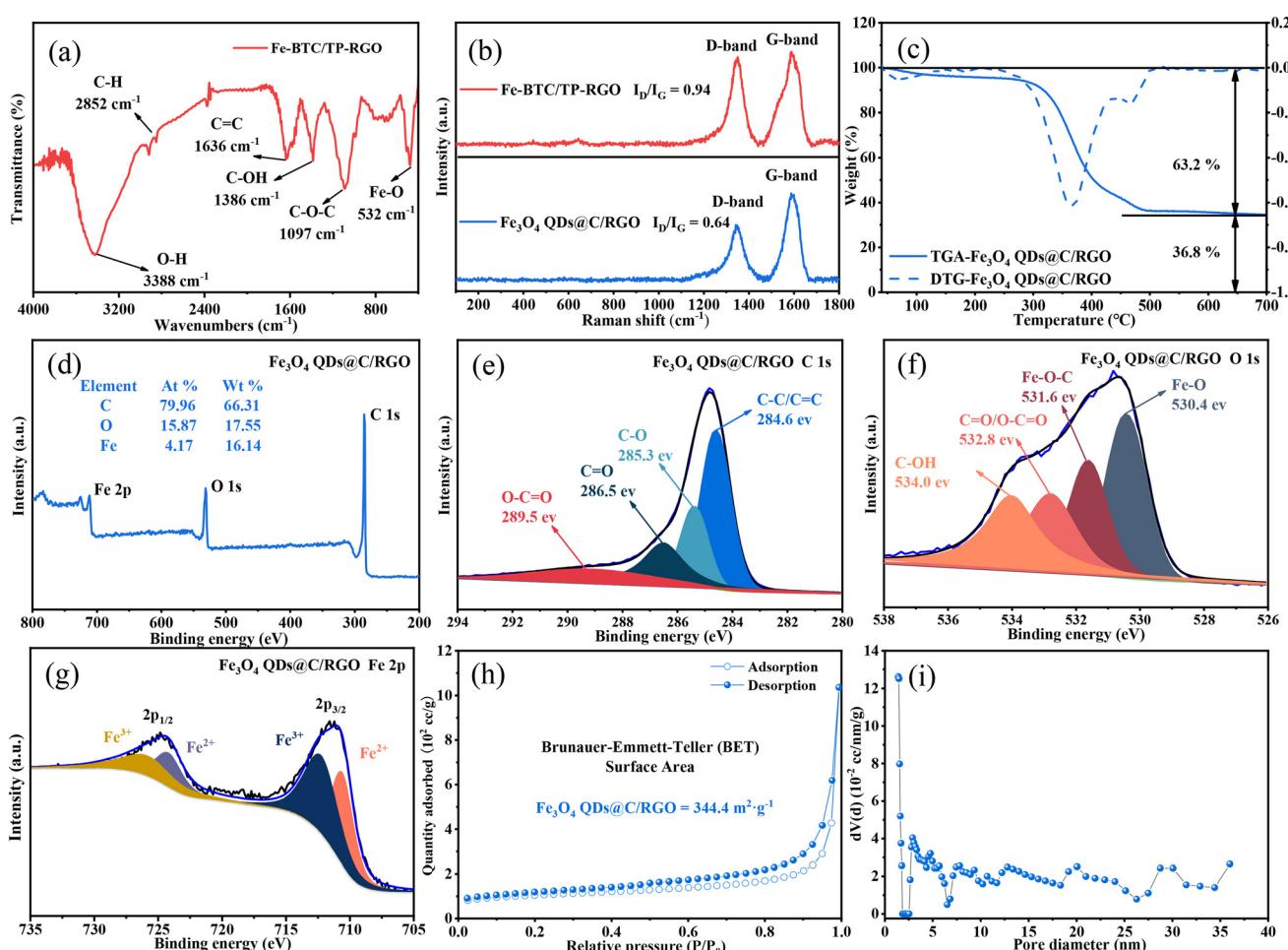


Fig. 2 FTIR pattern (a) of the Fe-BTC/TP-RGO sample. Raman spectra (b) of the Fe-BTC/TP-RGO and  $\text{Fe}_3\text{O}_4$  QDs@C/RGO samples. TGA and DTG curves (c) of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample. Survey XPS spectrum with atomic percentage summary (d). High-resolution C 1s (e), O 1s (f), Fe 2p (g) XPS spectra for the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample.  $\text{N}_2$  adsorption and desorption isotherms (h) and pore size distribution (i) of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample.

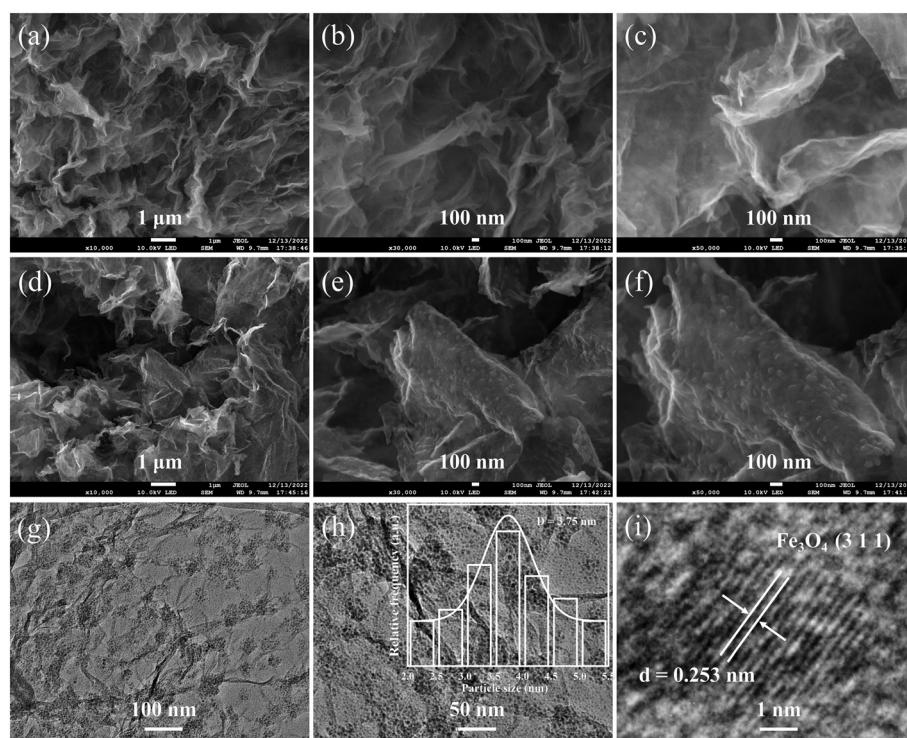


°C and 469.2 °C, respectively. The  $\text{Fe}_2\text{O}_3$  remnant is stabilized at 36.8%. Therefore, the corresponding  $\text{Fe}_3\text{O}_4$  QDs can be calculated to be 35.6% by mass while the carbon content is 64.4% in the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite.

The chemical valence and composition can be determined using the XPS technique. Fig. 2d shows the survey spectrum of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite, which consists of the three elements C, O, and Fe with corresponding weight percentages of 66.31%, 17.55%, and 16.14%, respectively. This result reveals the high purity of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite. As shown in Fig. 2e, the high-resolution C 1s spectrum can be resolved into four fitted peaks at 284.6, 285.3, 286.5, and 289.5 eV, which are attributed to C-C/C=C, C-O, C=O, and O-C=O bonds, respectively.<sup>39,40</sup> From the magnified O 1s spectrum in Fig. 2f, the peaks at 530.4, 531.5, and 533.3 eV are ascribed to Fe-O, Fe-O-C, and C-O/C=O bonds, respectively. The strong interaction between  $\text{Fe}_3\text{O}_4$  and carbon matrix *via* the Fe-O-C linkage will be beneficial to maintain the structural stability of  $\text{Fe}_3\text{O}_4$  QDs@C/RGO in the processes of discharging and charging as an anode material for LIBs.<sup>41,42</sup> Fig. 2g exhibits the high-resolution Fe 2p spectrum with two peaks located at 724.6 and 711.0 eV. For the fitting curves, two dominant peaks located at 726.0 and 712.4 eV are ascribed to the Fe 2p<sub>1/2</sub> and Fe 2p<sub>3/2</sub> states of Fe(III), while the other two peaks at 724.4 and 710.8 eV correspond to the Fe 2p<sub>1/2</sub> and Fe 2p<sub>3/2</sub> states of Fe(II), respectively.

To obtain the pore size distribution information and specific surface area of  $\text{Fe}_3\text{O}_4$  QDs@C/RGO samples, nitrogen adsorption and desorption isotherms have been recorded. As shown in Fig. 2h, the isotherms are of type IV with a hysteresis loop between 0.4 and 1.0 relative pressure, indicating the mesoporous nature of the sample.<sup>43,44</sup> In addition, the BET specific surface area of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample is estimated to be  $344 \text{ m}^2 \text{ g}^{-1}$ . The larger specific surface area facilitates the contact between electrolyte and active material.<sup>45,46</sup> As seen from BJH pore-size distribution curves in Fig. 2i, the overwhelming existence of mesopores with a wide range of pore diameters for the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite is verified.

FESEM images at different magnifications are used to investigate the morphological changes from the Fe-BTC/TP-RGO precursor to the final  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample. From the FESEM images of the Fe-BTC/TP-RGO precursor in Fig. 3a-c, continuous large-sized RGO framework with corrugated microstructures can be clearly seen. Moreover, no obvious restacking and the smooth surface of the RGO sheets can also be verified. These results confirm the significantly controlled crystallization of the Fe-BTC domains on the surface of the RGO sheets, which is favorable for the engineering of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite. After thermal treatment, the thin and wrinkled RGO framework can also be seen in Fig. 3d. In Fig. 3e and f, well-distributed small clusters are uniformly dispersed and anchored on the



**Fig. 3** FESEM images of Fe-BTC/TP-RGO (a–c) and  $\text{Fe}_3\text{O}_4$  QDs@C/RGO (d–f) samples under various magnifications. TEM images (g and h) of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample under different magnifications with inset showing  $\text{Fe}_3\text{O}_4$  particle size distribution. HRTEM image (i) for an individual  $\text{Fe}_3\text{O}_4$  nanoparticle in the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample.



RGO layer, which is highly related to the pyrolysis of the decorated Fe-BTC domains and induced size growth for the corresponding  $\text{Fe}_3\text{O}_4@\text{C}$  domains. From the TEM images of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample in Fig. 3g and h, small  $\text{Fe}_3\text{O}_4$  QDs locally assembled in a cluster configuration can be observed on the surface of the wrinkled RGO sheets. Most of the  $\text{Fe}_3\text{O}_4$  QD-containing clusters are below 100 nm, and obvious space between the  $\text{Fe}_3\text{O}_4$  QDs can also be verified. The particle size of  $\text{Fe}_3\text{O}_4$  is accurately measured and the data nonlinearly fitted by the Gaussian algorithm, as shown in the inset image of Fig. 3h. The particle sizes are in the range of 2.0–5.5 nm, and the corresponding average particle size is calculated to be 3.75 nm. From the HRTEM image in Fig. 3i, the crystallographic spacing of an individual  $\text{Fe}_3\text{O}_4$  QD is measured to be 0.253 nm, matching the (3 1 1) crystal plane of the  $\text{Fe}_3\text{O}_4$  species.

### 3.2 Electrochemical characterization

Fig. 4a exhibits the CV profiles of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite electrode for an initial five cycles with a scan rate of  $0.1 \text{ mV s}^{-1}$  in the potential range of 0.01 to 3.0 V (vs.  $\text{Li}^+/\text{Li}$ ). In the curve of the initial cathodic scanning, a sharp peak at about 0.64 V can be observed, which corresponds to the reduction of  $\text{Fe}_3\text{O}_4$  nanocrystals to metal Fe phase, accompanied by the generation of irreversible solid electrolyte interface (SEI) and the side reactions of electrolyte decomposition.<sup>47,48</sup> The obvious peak at about 0.01 V can be ascribed to the  $\text{Li}^+$  insertion into the carbon matrix.<sup>49,50</sup> In the subsequent curves, the cathodic peak shifts slightly to a higher voltage of about 0.81 V, which can be ascribed to the enhanced polarization and structural modification of

electrode material caused by  $\text{Li}^+$  insertion-extraction during the initial cycle.<sup>51,52</sup> Moreover, the two anodic peaks at 1.59 and 1.88 V may originate from the oxidation of the metallic Fe phase.<sup>53</sup> It is noteworthy that the subsequent CV curves almost overlap, demonstrating the good reversibility of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite electrode.

Fig. 4b shows the profiles of the constant current charge and discharge tests of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite at  $200 \text{ mA g}^{-1}$ . The initial cycle provides charge and discharge capacities of about 1266.62 and  $2098.59 \text{ mA h g}^{-1}$ , respectively. Thus, the initial coulombic efficiency can be calculated to be 60.36%. The reason for the low initial coulombic efficiency is most likely caused by the formation of SEI.<sup>54,55</sup> The voltage plateaus for this electrode are in good agreement with the redox peaks in the CV curves. Fig. 4c shows the cycling performance of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode at a low-density current of  $200 \text{ mA g}^{-1}$ . The specific capacity is maintained at  $1070.84 \text{ mA h g}^{-1}$  after 150 cycles. Fig. 4d shows the long-time lithium-ion storage capability of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode at a high-density current of  $1000 \text{ mA g}^{-1}$ . At the 300th cycle, this electrode still provides a high reversible capacity of about  $738.03 \text{ mA h g}^{-1}$ . These results indicate the good stability of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode. Due to the formation of metallic iron, this electrode exhibits a gradual capacity fading in the first few cycles. At low potentials, the *in situ* formed iron atoms promote the generation of a conducting polymer film, which contributes to the increase of lithium-ion storage *via* a pseudocapacitive behavior. Accompanying the improved reaction conditions, metallic iron atoms gradually participate in reversible electrochemical reactions.<sup>56,57</sup> The above factors may

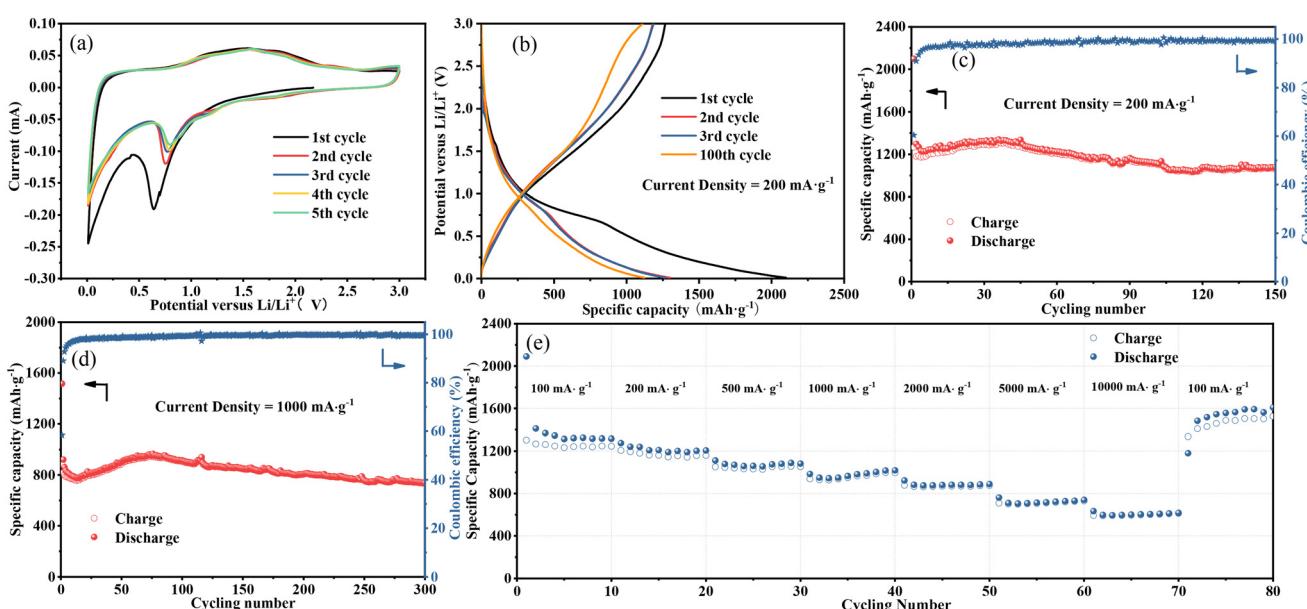


Fig. 4 The CV curves for the initial 5 cycles (a), and charge and discharge voltage profile (b) of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite. Cycling performance under a low current density of  $200 \text{ mA g}^{-1}$  (c), long-term lithium-ion storage capability under a current density of  $1000 \text{ mA g}^{-1}$  (d), and rate capability (e) of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite.



account for a continuous capacity increase in the following cycles.

The rate capability performance of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode is shown in Fig. 4e. The average reversible capacities at current densities of 100, 200, 500, 1000, 2000, 5000, and 10 000  $\text{mA g}^{-1}$  are 1336.84, 1216.56, 1076.31, 980.17, 886.00, 723.83, and 607.12  $\text{mA h g}^{-1}$ , respectively. The electrode provides a high reversible capacity of about 1179.44  $\text{mA h g}^{-1}$  when the current density is restored to the initial current density of 100  $\text{mA g}^{-1}$ , and the capacity is gradually increased in subsequent cycles. The unique structure with  $\text{Fe}_3\text{O}_4$  QDs immobilized by pyrolytic carbon and RGO skeleton offers sufficient electric and ionic pathways at high current rate and maintains superior structural stability. Therefore, the extraordinary rate capability of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode can be achieved.

A brief comparison of the synthesis method, reversible capacity, cycle life, and current density of  $\text{Fe}_3\text{O}_4$ -based anode materials is listed in Table 1.<sup>58–65</sup> The  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite electrode has the obvious merits of high reversible capacity, excellent long-time cycling stability as well as a facile synthesis method.

As presented in Fig. 5a, the dynamic behavior at different scan rates was tested to further investigate the lithium storage mechanism and diffusion kinetics of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode. The seven CV curves maintain similar properties and the peak current increases when experiencing a gradual increase in scan rate. These results indicate that the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode has excellent electrochemical reversibility performance. The electrochemical reaction can be evaluated according to the relationship between peak current ( $i$ ) and scanning rate ( $v$ ), which follows the power-law relationship:  $i = av^b$ , where  $a$  and  $b$  are fitting parameters. The value of  $b$  varies between 0.5 (diffusion-controlled process) and 1.0 (capacitive-controlled process), which is obtained from the slope of a  $\ln(i)$ - $\ln(v)$  plot.<sup>66,67</sup> As shown in Fig. 5b, the  $b$  values of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode can be calculated to be 0.714 and 0.855 for the cathode and anode peaks, which indicates the coexistence of diffusion-controlled processes and pseudocapacitive processes during the electrochemical

reaction. The reversible capacity contribution can be calculated by the formula:  $i = k_1v + k_2v^{1/2}$ , where  $k_1v$  and  $k_2v^{1/2}$  correspond to the pseudocapacitive contribution and diffusion-controlled contribution, respectively.<sup>68,69</sup> Fig. 5c shows the results of quantitatively analyzing the capacitive contribution of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode. At a scan rate of 1.0  $\text{mV s}^{-1}$ , the pseudocapacitive contribution rate illustrated by the pink region is about 75% of the total lithium-ion storage capacity. In Fig. 5d, with the increase of scan rate from 0.1  $\text{mV s}^{-1}$  to 1.0  $\text{mV s}^{-1}$ , the pseudocapacitive contribution rate increases to 49.0%, 55.3%, 59.6%, 63.0%, 65.3%, 71.4%, and 75.0%, respectively. The pseudocapacitive behavior of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode is another factor for the inspiring rate performance.<sup>70,71</sup>

The galvanostatic intermittent titration technique (GITT) is regarded as an efficient method to explore the  $\text{Li}^+$  diffusion coefficients ( $D_{\text{Li}^+}$ ) for the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite electrode. Fig. 6a presents the GITT curves during the first cycle (tested at 200  $\text{mA g}^{-1}$ , pulse time 20 min, relaxation time 30 min). According to Fick's second law of diffusion,

$$D_{\text{Li}^+} = \frac{4}{\pi\tau} \left( \frac{m_B V_m}{M_B A} \right)^2 \left( \frac{\Delta E_S}{\Delta E_\tau} \right)^2, \text{ in}$$

which  $\tau$  is relaxation time (min),  $m_B$ ,  $V_m$ , and  $M_B$  stand for the mass, molar volume and molecular weight of the electrode material, respectively,  $A$  represents the area of the electrode plate,  $\Delta E_S$  is the steady-state potential change via the current pulse, and  $\Delta E_\tau$  is the potential change in the current pulse after subtracting the IR drop.<sup>72–74</sup> Fig. 6b illustrates a magnified single GITT curve, which is specifically marked with the parameters of  $\tau$ ,  $\Delta E_S$  and  $\Delta E_\tau$ . As shown in Fig. 6c and d, the average value of  $D_{\text{Li}^+}$  reached  $2.4 \times 10^{-8} \text{ cm}^2 \text{ s}^{-1}$  and  $2.6 \times 10^{-8} \text{ cm}^2 \text{ s}^{-1}$  during the lithiation and delithiation processes, respectively. This result implies that an optimized  $\text{Li}^+$  diffusion condition has been created in the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite.

FESEM images of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode before and after 100 cycles are employed to investigate the morphological and structural evolution. From Fig. 7a and b, it can be seen that the overall microstructure of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite has been largely maintained after electrode preparation, indicating the good mechanical

**Table 1** Comparison of the lithium-ion storage performance of carbon materials and carbonaceous matrix-supported  $\text{Fe}_3\text{O}_4$  materials

Sample name	Synthesis method	Reversible capacity ( $\text{mA h g}^{-1}$ )	Cycle life	Current density ( $\text{mA g}^{-1}$ )	Year published
$\text{Fe}_3\text{O}_4$ @C-500 (ref. 58)	Carbonization	718 190	500 2000	200 1000	2023
$\text{Fe}_3\text{O}_4$ @HCS <sup>59</sup>	Hydrothermal carbonization	1050	250	100	2023
$\text{Fe}_3\text{O}_4$ /C (ref. 60)	Carbonization	405.6	50	100	2023
Si-QDs/ $\text{Fe}_3\text{O}_4$ /rGO <sup>61</sup>	Self-assembly annealing	1367.1	80	100	2023
$\text{Fe}_3\text{O}_4$ @void@N-doped C-5 (ref. 62)	Hydrothermal carbonization	1222	100	200	2022
$\text{Fe}_3\text{O}_4$ @C (ref. 63)	Hydrothermal carbonization	291.7	300	1000	2022
PTA-700 (ref. 64)	Hydrothermal carbonization	535	100	100	2022
$\text{Fe}_3\text{O}_4$ @CTP QDs <sup>65</sup>	Coprecipitation	810	200	100	2022
$\text{Fe}_3\text{O}_4$ QDs@C/RGO	Hydrothermal calcination	<b>738.03</b> <b>1070.84</b>	<b>300</b> <b>150</b>	<b>1000</b> <b>200</b>	<b>This work</b>



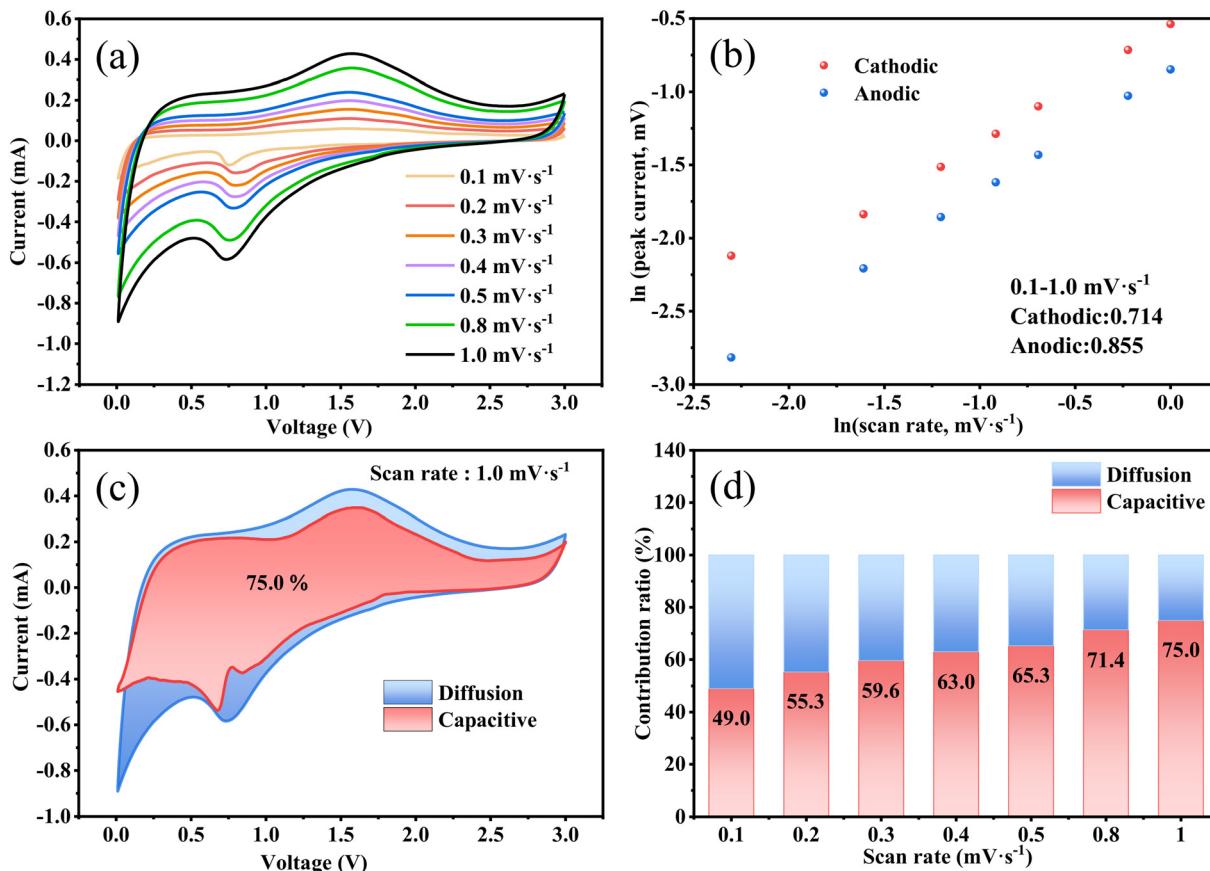


Fig. 5 CV curves of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite electrode from 0.1 to 1.0  $\text{mV}\cdot\text{s}^{-1}$  (a), the  $\ln(i)$ - $\ln(v)$  plot (b), the capacitive contribution of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite electrode at 1  $\text{mV}\cdot\text{s}^{-1}$  (c) and the pseudocapacitive contribution ratios at different scanning rates (d).

strength of this composite sample. From Fig. 7c and d, an apparent smooth SEI film could be observed on the surface of the electrode after 100 cycles, while the electrode still maintains its integrity. These results well explain the good structural stability and excellent lithium storage properties of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode. EIS provides comprehensive interfacial electrochemical information, especially in the study of kinetic reaction mechanisms and the measurement of kinetic parameters of LIBs. The corresponding Nyquist plots of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode at different states are shown in Fig. 7e. In the equivalent circuit, constant phase angle element (CPE) is widely used in the equivalent circuit of alternating-current impedance to fit the experimental impedance data. The SEI film resistance ( $R_s$ ) is the intersection of the curve starting point and the horizontal axis in the high-frequency region.<sup>75</sup> The charge transfer resistance ( $R_{ct}$ ) is related to the diameter of the semicircle in the high-frequency and mid-frequency regions.<sup>76</sup> Warburg impedance ( $Z_w$ ) corresponds to the slope of the line in the low-frequency region, reflecting the  $\text{Li}^+$  diffusion in the electrode material.<sup>77</sup> According to the fitting results, it can be observed that the semicircle diameter obtained after 100 cycles for the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite electrode in the high- to medium-frequency region is significantly

smaller than that before the cycling test, which illustrates the stable SEI formation at low potentials, better contact between the electrolyte and active material, and a significant reduction of charge transfer resistance after the activation of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO composite.<sup>78</sup>

With a view to the practicality of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO sample, full cells are configurated by integrating an  $\text{Fe}_3\text{O}_4$  QDs@C/RGO anode and a commercial  $\text{LiFePO}_4$  cathode. To guarantee the excellent electrochemical performance of full cells, the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO anode is electrochemically pre-lithiated to compensate for  $\text{Li}^+$  consumption in the initial lithiation process before fabricating the full cells. Fig. 8a illustrates the charge/discharge profiles of the  $\text{LiFePO}_4$  cathode material under a low current density of 200  $\text{mA g}^{-1}$ , which exhibits a reversible plateau around 3.3 V. As shown in Fig. 8b, the reversible capacity reaches approximately 116.36  $\text{mA h g}^{-1}$  after 100 cycles. Fig. 8c shows the galvanostatic charge/discharge voltage profiles of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO// $\text{LiFePO}_4$  full cell between 0.01 and 3.7 V at a current density at 200  $\text{mA g}^{-1}$ . The full cell exhibits initial lithiation and delithiation capacities of 1597.83 and 1268.54  $\text{mA h g}^{-1}$ , and the corresponding coulombic efficiency is calculated to be 79.39%. Moreover, the shape of the curves for the subsequent cycles is largely maintained compared with the first cycle, indicating



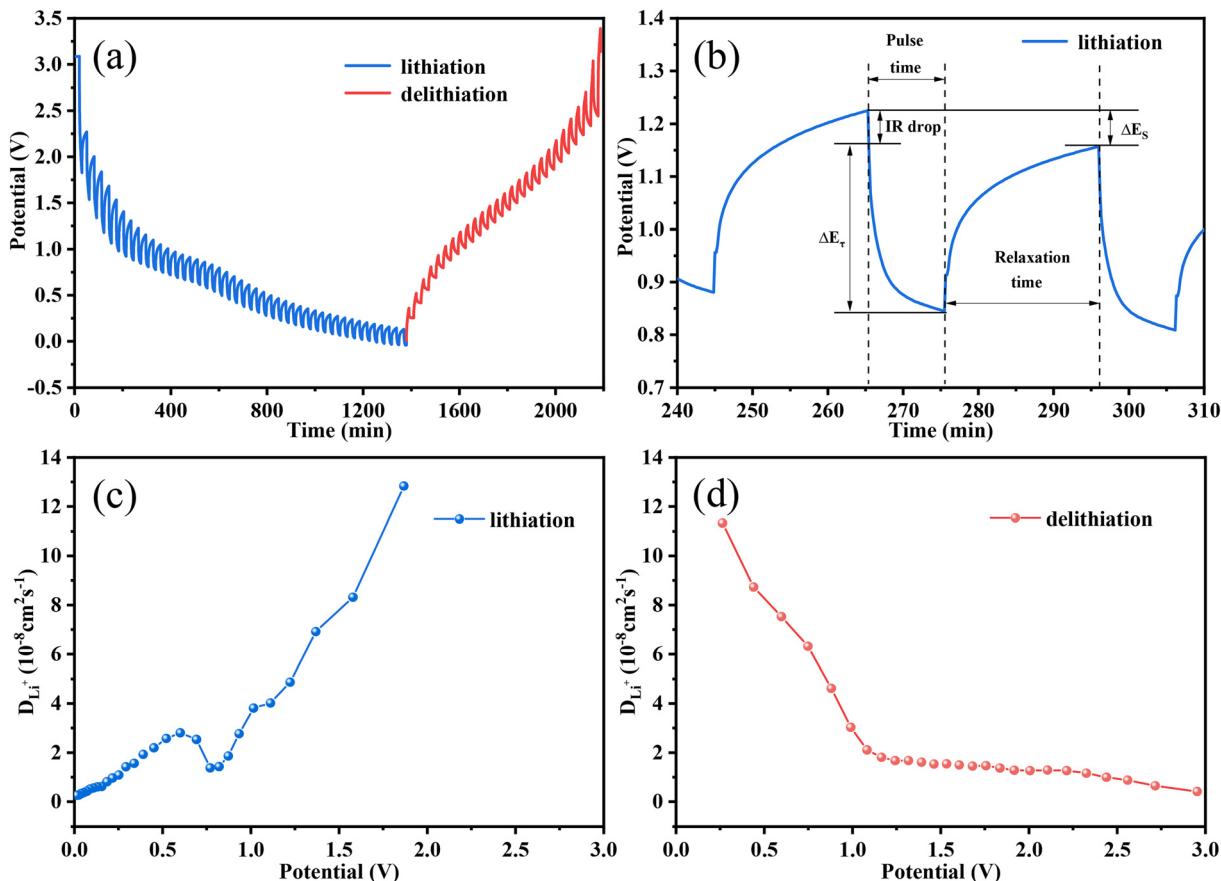


Fig. 6 GITT curves (a) and a profile for a single GITT test (b), and variation of the  $D_{Li^+}$  values during lithiation (c) and delithiation (d) processes of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode.

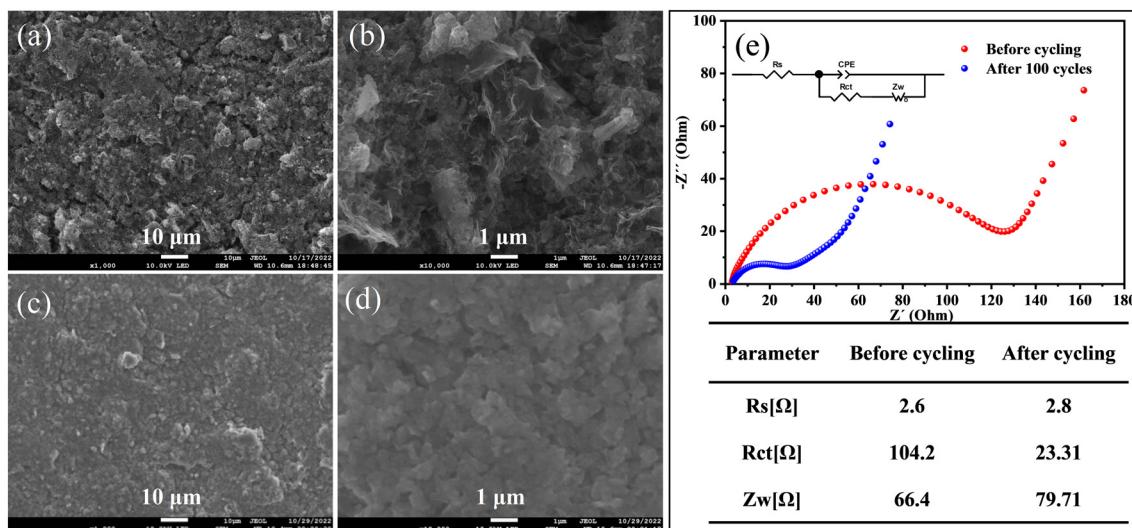
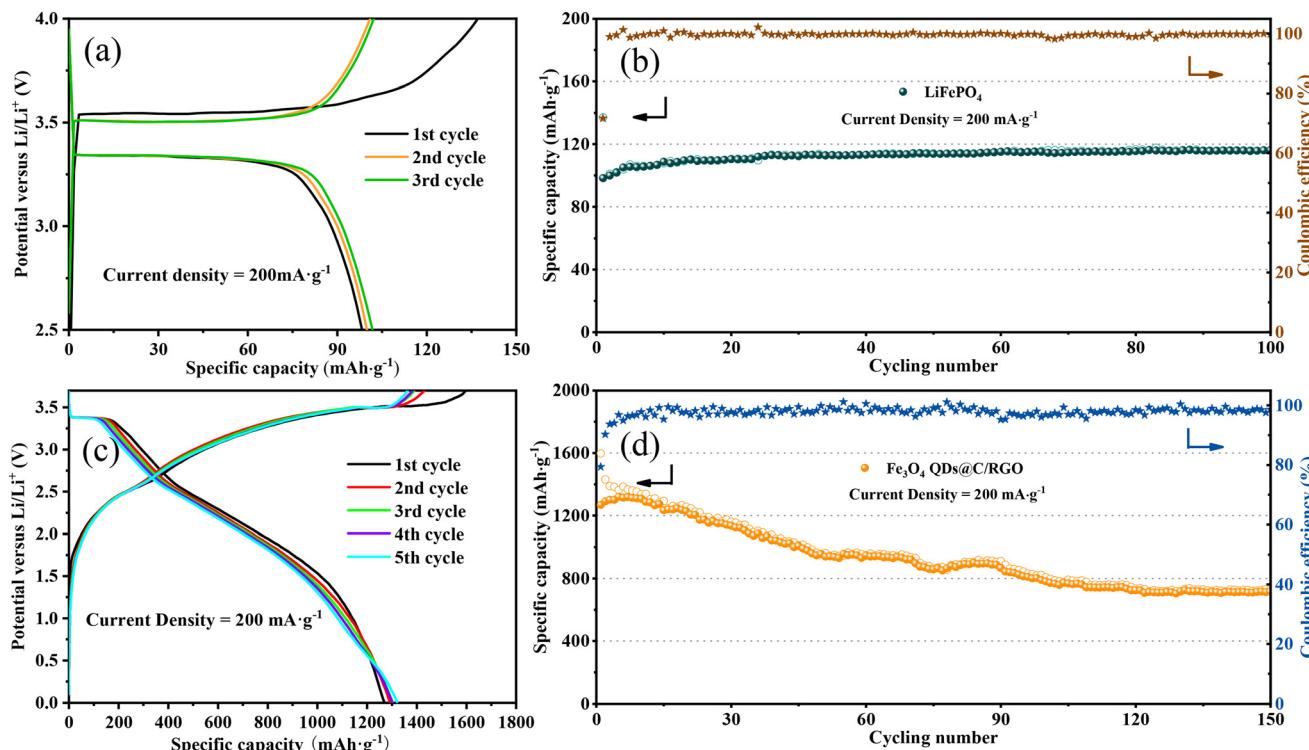


Fig. 7 FESEM images of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode before cycling (a and b) and after 100 cycles (c and d) under various magnifications. Nyquist plots of the  $\text{Fe}_3\text{O}_4$  QDs@C/RGO electrode before cycling and after 100 cycles with inset showing equivalent circuit and the corresponding fitted values of the parameters (e).

excellent structural stability for the composite over long-time cycling. From Fig. 8d, the full cell still exhibits a high reversible capacity of about  $717.29 \text{ mA h g}^{-1}$  after 150 cycles under a current density of  $200 \text{ mA g}^{-1}$ .

## 4. Conclusions

In a mild aqueous reaction system containing  $\text{H}_3\text{BTC}$ , TP, GO, and iron foils, an Fe-BTC/TP-RGO precursor sample with



**Fig. 8** Charge and discharge voltage profile (a) and cycling performance (b) of a commercial LiFePO<sub>4</sub> cathode at 200 mA g<sup>-1</sup> in half cells. Charge and discharge voltage profile (c) and cycling performance (d) of the Fe<sub>3</sub>O<sub>4</sub> QDs@C/RGO sample in Fe<sub>3</sub>O<sub>4</sub> QDs@C/RGO//LiFePO<sub>4</sub> full cells at 200 mA g<sup>-1</sup>.

improved microstructure is rationally engineered. Due to the modulation of the TP molecules, the spontaneous restacking of the RGO sheets has been effectively inhibited and the size growth of the Fe-BTC MOF domains has been significantly controlled. After a subsequent thermal treatment, the Fe<sub>3</sub>O<sub>4</sub> QDs@C/RGO composite material is engineered, which has a unique structure with Fe<sub>3</sub>O<sub>4</sub> QDs immobilized by pyrolytic carbon and RGO skeleton. The Fe<sub>3</sub>O<sub>4</sub> QDs@C/RGO composite electrode exhibits a more attractive lithium-ion storage capacity with a reversible capacity of 1070.84 mA h g<sup>-1</sup> for the electrode after 100 cycles at a low-density current of 200 mA g<sup>-1</sup>, and a high reversible capacity of about 738.03 mA h g<sup>-1</sup> after the 300th cycle at a high-density current of 1000 mA g<sup>-1</sup>. Moreover, the Fe<sub>3</sub>O<sub>4</sub> QDs@C/RGO sample could still have a high discharge capacity of 717.29 mA h g<sup>-1</sup> after 150 cycles in Fe<sub>3</sub>O<sub>4</sub> QDs@C/RGO//LiFePO<sub>4</sub> full cells between 0.01 and 3.7 V at 200 mA g<sup>-1</sup>. The sample synthesis protocol is facile and eco-friendly, and the synthesized Fe<sub>3</sub>O<sub>4</sub> QDs@C/RGO sample has good potential in wider fields.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

This work was financially supported by the Thousand Talents Plan, the World-Class University and Discipline, the Taishan

Scholar's Advantageous and Distinctive Discipline Program and the World-Class Discipline Program of Shandong Province.

## References

1. F. Wu, J. Maier and Y. Yu, *Chem. Soc. Rev.*, 2020, **49**, 1569–1614.
2. J. Xu, X. Cai, S. Cai, Y. Shao, C. Hu, S. Lu and S. Ding, *Energy Environ. Sci.*, 2023, **0**, e12450.
3. J. Li, J. Fleetwood, W. B. Hawley and W. Kays, *Chem. Rev.*, 2022, **122**, 903–956.
4. L. Zhang, C. Zhu, S. Yu, D. Ge and H. Zhou, *J. Energy Chem.*, 2022, **66**, 260–294.
5. H. Cheng, J. G. Shapter, Y. Li and G. Gao, *J. Energy Chem.*, 2021, **57**, 451–468.
6. C. Wang, C. Yang and Z. Zheng, *Adv. Sci.*, 2022, **9**, e2105213.
7. C. Hu, Y. J. Hu, A. P. Chen, X. Z. Duan, H. Jiang and C. Z. Li, *Engineering*, 2022, **18**, 154–160.
8. Y. Liu, C. Hu, L. Chen, Y. J. Hu, H. Jiang and C. Z. Li, *J. Energy Chem.*, 2022, **69**, 450–455.
9. C. Hu, L. Chen, Y. J. Hu, A. P. Chen, L. Chen, H. Jiang and C. Z. Li, *Adv. Mater.*, 2021, **33**, 2103558.
10. J. P. Xie, J. L. Li, X. D. Li, H. Lei, W. C. Zhuo, X. B. Li, G. Hong, K. N. Hui, L. K. Pan and W. J. Mai, *CCS Chem.*, 2021, **3**, 791–799.
11. Z. Yi, D. L. Fang, W. Q. Zhang, J. Tian, S. M. Chen, J. B. Liang, N. Lin and Y. T. Qian, *CCS Chem.*, 2021, **3**, 1306–1315.



12 N. Wang, Y. Y. Liu, Z. X. Shi, Z. L. Yu, H. Y. Duan, S. Fang, J. Y. Yang and X. M. Wang, *Rare Met.*, 2021, **41**, 438–447.

13 S. Fang, D. Bresser and S. Passerini, *Adv. Energy Mater.*, 2019, **10**, 1902485.

14 Y. Yang, W. Yuan, X. Zhang, C. Wang, Y. Yuan, Y. Huang, Y. Ye, Z. Qiu and Y. Tang, *Renewable Sustainable Energy Rev.*, 2020, **127**, 109884.

15 R. Puttaswamy, R. K. Pai and D. Ghosh, *J. Mater. Chem. A*, 2022, **10**, 508–553.

16 Y. Yu, T. Ma and H. Huang, *Adv. Funct. Mater.*, 2023, **33**, 2213770.

17 J. F. S. Fernando, C. Zhang, K. L. Firestein, J. Y. Nerkar and D. V. Golberg, *J. Mater. Chem. A*, 2019, **7**, 8460–8471.

18 C. X. Peng, B. D. Chen, Y. Qin, S. H. Yang, C. Z. Li, Y. H. Zuo, S. Y. Liu and J. H. Yang, *ACS Nano*, 2012, **6**, 1074–1081.

19 J. Y. Wu, H. Lin, D. J. Moss, K. P. Loh and B. H. Jia, *Nat. Rev. Chem.*, 2023, **7**, 162–183.

20 F. Wang, W. Fang, X. Ming, Y. Liu, Z. Xu and C. Gao, *Appl. Phys. Rev.*, 2023, **10**, 011311.

21 R. C. K. Reddy, J. Lin, Y. Chen, C. Zeng, X. Lin, Y. Cai and C.-Y. Su, *Coord. Chem. Rev.*, 2020, **420**, 213434.

22 R. A. Bajwa, U. Farooq, S. Ullah, M. Salman, S. Haider and R. Hussain, *J. Energy Storage*, 2023, **72**, 108708.

23 M. H. Shen and H. L. Ma, *Coord. Chem. Rev.*, 2022, **470**, 214715.

24 Y. Zhang, Y. F. Liu, D. Wang, J. C. Liu, J. W. Zhao and L. J. Chen, *Polyoxometalates*, 2023, **2**, 9140017.

25 Y. An, L. L. Wang, W. Y. Jiang, X. L. Lv, G. Q. Yuan, X. X. Hang and H. Pang, *Polyoxometalates*, 2023, **2**, 9140030.

26 R. Jia, R. Zhang, L. Yu, X. Kong, S. Bao, M. Tu, X. Liu and B. Xu, *J. Colloid Interface Sci.*, 2023, **630**, 86–98.

27 Y. Wang, Z. Shi and J. Yin, *ACS Appl. Mater. Interfaces*, 2011, **3**, 1127–1133.

28 O. Akhavan, M. Kalaei, Z. S. Alavi, S. M. A. Ghiasi and A. Esfandiar, *Carbon*, 2012, **50**, 3015–3025.

29 X. Weng, J. Wu, L. Ma, G. Owens and Z. Chen, *Chem. Eng. J.*, 2019, **359**, 976–981.

30 B. Xu, J. Zhang, Y. Gu, Z. Zhang, W. Al Abdulla, N. A. Kumar and X. S. Zhao, *Electrochim. Acta*, 2016, **212**, 473–480.

31 Y. Yang, Y. Bai, F. Zhao, E. Yao, J. Yi, C. Xuan and S. Chen, *RSC Adv.*, 2016, **6**, 67308–67314.

32 T. A. Vu, G. H. Le, H. T. Vu, K. T. Nguyen, T. T. T. Quan, Q. K. Nguyen, H. T. K. Tran, P. T. Dang, L. D. Vu and G. D. Lee, *Mater. Res. Express*, 2017, **4**, 035038.

33 H. T. Vu, M. B. Nguyen, T. M. Vu, G. H. Le, T. T. T. Pham, T. D. Nguyen and T. A. Vu, *Top. Catal.*, 2020, **63**, 1046–1055.

34 B. Xu, L. Yu, X. Zhao, H. Wang, C. Wang, L. Y. Zhang and G. Wu, *J. Colloid Interface Sci.*, 2021, **584**, 827–837.

35 C. Xiao, H. Li, Y. Zhao, X. Zhang and X. Wang, *J. Environ. Manage.*, 2020, **275**, 111262.

36 X. Kong, L. Shan, R. Zhang, S. Bao, M. Tu, R. Jia, L. Yu, H. Li and B. Xu, *J. Colloid Interface Sci.*, 2022, **628**, 1–13.

37 P. Pachfule, D. Shinde, M. Majumder and Q. Xu, *Nat. Chem.*, 2016, **8**, 718–724.

38 D. Graf, F. Molitor, K. Ensslin, C. Stampfer, A. Jungen, C. Hierold and L. Wirtz, *Nano Lett.*, 2007, **7**, 238–242.

39 Z. Shen, H. Xing, Y. Zhu, X. Ji, Z. Liu and L. Wang, *J. Mater. Sci.: Mater. Electron.*, 2017, **28**, 13896–13904.

40 B. Xu, X. Guan, L. Y. Zhang, X. Liu, Z. Jiao, X. Liu, X. Hu and X. S. Zhao, *J. Mater. Chem. A*, 2018, **6**, 4048–4054.

41 Y. Cui, W. Feng, W. Liu, J. Li, Y. Zhang, Y. Du, M. Li, W. Huang, H. Wang and S. Liu, *Nanoscale*, 2020, **12**, 10816–10826.

42 Y. Liu, D. He, Q. Tan, Q. Wan, K. Han, Z. Liu, P. Li, F. An and X. Qu, *J. Mater. Chem. A*, 2019, **7**, 19430–19441.

43 Q. Wu, R. Yu, Z. Zhou, H. Liu and R. Jiang, *Langmuir*, 2021, **37**, 785–792.

44 Y. Yuan, Z. Kong, L. Qiao, Z. Xu, Z. Wang, X. Teng, Y. Dong, X. Liu, A. Fu, Y. Li and H. Li, *ChemElectroChem*, 2021, **8**, 4480–4489.

45 Y. Yin, C. Ma, W. D. Cai, W. M. Qiao, L. C. Ling and J. T. Wang, *ChemNanoMat*, 2022, **8**, e202100490.

46 X. Meng, Y. Xu, X. Sun, J. Wang, L. Xiong, X. Du and S. Mao, *J. Mater. Chem. A*, 2015, **3**, 12938–12946.

47 Y. He, L. Huang, J.-S. Cai, X.-M. Zheng and S.-G. Sun, *Electrochim. Acta*, 2010, **55**, 1140–1144.

48 Y. Wang, L. Chen, H. Liu, Z. Xiong, L. Zhao, S. Liu, C. Huang and Y. Zhao, *Chem. Eng. J.*, 2019, **356**, 746–755.

49 Y. Cui, W. Liu, Y. Lyu, Y. Zhang, H. Wang, Y. Liu and D. Li, *J. Mater. Chem. A*, 2018, **6**, 18276–18285.

50 J. Song, Y. Ji, Y. Li, X. Lu, W. Ren, Q. Tian, J. Chen and L. Yang, *Ceram. Int.*, 2021, **47**, 26092–26099.

51 Y. Ma, J. Huang, L. Lin, Q. Xie, M. Yan, B. Qu, L. Wang, L. Mai and D. Peng, *J. Power Sources*, 2017, **365**, 98–108.

52 J. Cheng, Y. Zhou, Y. Wu, Y. Pan, X. Luo, Y. Huang, Y. Zhou, L. Zhu and Z. Yuan, *J. Alloys Compd.*, 2023, **930**, 167429.

53 D. Yoon, J. Hwang, W. Chang and J. Kim, *Chem. Eng. J.*, 2017, **317**, 890–900.

54 R. Zhang, S. Bao, Q. Tan, B. Li, C. Wang, L. Shan, C. Wang and B. Xu, *J. Colloid Interface Sci.*, 2021, **600**, 602–612.

55 Y. Yan, X. Lu, Y. Li, J. Song, Q. Tian, L. Yang and Z. Sui, *J. Alloys Compd.*, 2022, **899**, 163342.

56 R. Zhang, C. Lv, S. Bao, J. Gao, Y. Xie, F. Zheng, X. Liu, Y. Wen and B. Xu, *J. Colloid Interface Sci.*, 2022, **628**, 154–165.

57 X. Meng, J. Huang, Y. Bian, H. Du, Y. Xu, S. Zhu, Q. Li, M. Chen and M.-C. Lin, *J. Solid State Chem.*, 2021, **303**, 122456.

58 M. Li, W. Ma, F. Tan, B. Yu, G. Cheng, H. Gao and Z. Zhang, *J. Power Sources*, 2023, **574**, 233146.

59 L. Singer, W. Kukulka, E. Thauer, N. Gräßler, A. Asyuda, M. Zharnikov, E. Mijowska and R. Klingeler, *Electrochim. Acta*, 2023, **448**, 142155.

60 W. Yodying, T. Sarakonsri, N. Ratsameetammajak, K. Khunpakdee, M. Haruta and T. Autthawong, *Crystals*, 2023, **13**, 280.

61 W. Hu, K. He, S. Wu, T. Chen, X. Yu, Y. Liang, M. Zheng, Y. Xiao, H. Dong, Y. Liu and H. Hu, *J. Alloys Compd.*, 2023, **943**, 168947.

62 J. Wang, Q. Hu, W. Hu, W. Zhu, Y. Wei, K. Pan, M. Zheng and H. Pang, *Molecules*, 2022, **27**, 396.

63 J. Huang, Q. Dai, C. Cui, H. Ren, X. Lu, Y. Hong and S. Woo Joo, *J. Electroanal. Chem.*, 2022, **918**, 116508.



64 Q. Wei, H. Zhu, S. Yu, G. Xu, J. Yin, J. Tong, T. Chen, X. He, P. Guo, H. Jiang, J. Li and Y. Wang, *Appl. Surf. Sci.*, 2023, **608**, 155093.

65 F. Shi, Q. Liu, Z. Jin, G. Huang, B. Xing, J. Jia and C. Zhang, *J. Alloys Compd.*, 2022, **890**, 161911.

66 V. Augustyn, J. Come, M. A. Lowe, J. W. Kim, P. L. Taberna, S. H. Tolbert, H. D. Abruna, P. Simon and B. Dunn, *Nat. Mater.*, 2013, **12**, 518–522.

67 Q. Chen, W. Zhong, J. Zhang, C. Gao, W. Liu, G. Li and M. Ren, *J. Alloys Compd.*, 2019, **772**, 557–564.

68 V. Augustyn, P. Simon and B. Dunn, *Energy Environ. Sci.*, 2014, **7**, 1597.

69 Z. Liu, L. Liu, Z. Zhao, J. He, S. Wang and C. Xiong, *Appl. Surf. Sci.*, 2020, **526**, 146639.

70 J. Song, Y. Li, X. Lu, W. Zhang, Y. Xiang, J. Chen and Q. Tian, *Appl. Surf. Sci.*, 2022, **604**, 154502.

71 B. Yin, X. Cao, A. Pan, Z. Luo, S. Dinesh, J. Lin, Y. Tang, S. Liang and G. Cao, *Adv. Sci.*, 2018, **5**, 1800829.

72 C. Lv, C. Lin and X. S. Zhao, *Adv. Energy Mater.*, 2021, **12**, 2102550.

73 H. Zhang, P. Zong, M. Chen, H. Jin, Y. Bai, S. Li, F. Ma, H. Xu and K. Lian, *ACS Nano*, 2019, **13**, 3054–3062.

74 Y. Cheng, S. Wang, L. Zhou, L. Chang, W. Liu, D. Yin, Z. Yi and L. Wang, *Small*, 2020, **16**, e2000681.

75 R. Ding, J. Zhang, J. Qi, Z. Li, C. Wang and M. Chen, *ACS Appl. Mater. Interfaces*, 2018, **10**, 13470–13478.

76 Y. Xie, Y. Qiu, L. Tian, T. Liu and X. Su, *J. Alloys Compd.*, 2022, **894**, 162384.

77 H. Li, J. Wang, Y. Li, Y. Zhao, Y. Tian, I. Kurmanbayeva and Z. Bakenov, *J. Electroanal. Chem.*, 2019, **847**, 113240.

78 Y. Zhao, X. Zhai, D. Yan, C. Ding, N. Wu, D. Su, Y. Zhao, H. Zhou, X. Zhao, J. Li and H. Jin, *Electrochim. Acta*, 2017, **243**, 18–25.

