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Fe₃O₄@GO magnetic nanocomposites protect mesenchymal stem cells and promote osteogenic differentiation of rat bone marrow mesenchymal stem cells†

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Fe₃O₄ nanoparticles (Fe₃O₄ NPs) are typical magnetic materials for bone tissue regeneration. However, the accompanying oxidative stress during the reaction process of Fe₃O₄ NPs and H₂O₂ in bone remodeling and disease may hinder their application. In order to reduce this side effect, we selected graphene oxide (GO) to modify Fe₃O₄ NPs. We showed that Fe₃O₄@GO magnetic nanocomposites (Fe₃O₄@GO MNCs) eliminated 30% of H₂O₂ in 3 h, and reduced the amount of ·OH, the intermediate product of the Fenton reaction. The cellular study demonstrated that Fe₃O₄@GO MNCs reduced the cell damage caused by reactive oxygen species (ROS) and improved the activity of mesenchymal stem cells (MSCs). Moreover, when the magnetic field and bone morphogenetic protein-2 (BMP2) delivered by Fe₃O₄@GO MNCs worked together, osteogenic differentiation of MSCs *in vitro* was well promoted.

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Introduction

Physical cues such as mechanical,¹ electrical,² and magnetic³ stimulation play a crucial role in manipulating the cell fate processes, such as proliferation,⁴ differentiation of cells,⁵ as well as the permeability of ions across the cell membrane.^{6,7} As early as the last century, static magnetic field (SMF) had been applied to orthodontic treatments,⁸ demonstrating the clinical feasibility of magnetic stimulation.^{9,10} Over the decades, magnetic fields in the form of pulse electromagnetic field (PEF) and SMF have been proven to be effective in bone repair and regeneration^{11,12} because of their ability to promote the osteogenesis of osteoblasts during bone formation.^{13,14}

Exploration of clinical application along with inspiring progress in biomagnetism encouraged researchers to develop magnetic materials as carriers to load stem cells or growth factors for tissue engineering.¹⁵ In this regard, some of us recently demonstrated that in the presence of SMF, magnetic electrospun fibrous scaffolds with iron-doped hydroxyapatite not only promoted the proliferation and osteogenic differentiation of loaded mesenchymal stem cells (MSCs) *in vitro*, but also accelerated bone repair and new bone formation *in vivo*.¹⁶ On the other hand, numerous studies revealed a harsh condition for bone tissue regeneration in which the microenvironment was accompanied by the imbalance of reactive oxygen species (ROS).^{17,18} Therefore, it is necessary to develop a magnetic responsive scaffold/carrier with ROS modulating capability to mitigate locally excessive ROS and maintain the ability to deliver growth factors simultaneously, which would be beneficial to stem cell protection and bone tissue regeneration.

ROS is a family of chemical species with strong oxidizing ability. Important biological molecules such as lipid,¹⁹ protein,²⁰ and DNA²¹ can be damaged by ROS, resulting in a decreased survival rate of endogenous/transplanted stem cells and a lowered delivery efficiency of growth factors (e.g., bone morphogenetic protein-2, BMP2).^{22,23} Meanwhile, the overproduction of ROS is likely to induce inflammatory reactions and tissue injury,²⁴ which are interrelated in multiple diseases.^{25–27} Therefore, stabilizing ROS is a prerequisite for bone tissue repair.²⁸ Otherwise, abnormally high oxidative stress triggers cell apoptosis at injured tissues, leading to poor survival of the

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endogenous/transplanted stem cells or growth factors and finally ending up with low therapeutic efficacy.^{29,30} On the other hand, a low level of ROS at damaged tissues is beneficial to osteogenic rather than adipogenic differentiation of MSCs.³¹ Therefore, it is of clinical importance to delicately modulate the ROS level for treating bone diseases.³²

Iron oxide nanoparticles (*i.e.*, Fe₃O₄ NPs) as a classical magnetic material draw great attention since they have been successfully approved in magnetic resonance imaging by the Food and Drug Administration (FDA).^{33,34} Also, recent studies showed that Fe₃O₄ NPs exhibit an intrinsic peroxidase-like activity.^{35,36} This offers a possibility for Fe₃O₄ NPs to play a dual role in both ROS regulation and magnetic response. However, due to the presence of ferrous ions, Fe₃O₄ NPs underwent a Fenton like pathway which converted hydrogen peroxide into higher active $\cdot\text{OH}$.³⁷ The application of Fe₃O₄ NPs in oxidative stress environment would be limited by the highly reactive hydroxyl radical formed from the Fenton reaction.^{38,39} To mitigate the disadvantage of Fe₃O₄ NPs when applied in complex lesion,⁴⁰ GO was chosen to capture unwilling $\cdot\text{OH}$ from the Fenton reaction^{41,42} and to provide cytoprotection.⁴³ In addition, the large surface to volume ratio endows GO additional drug delivery capacity.^{44,45}

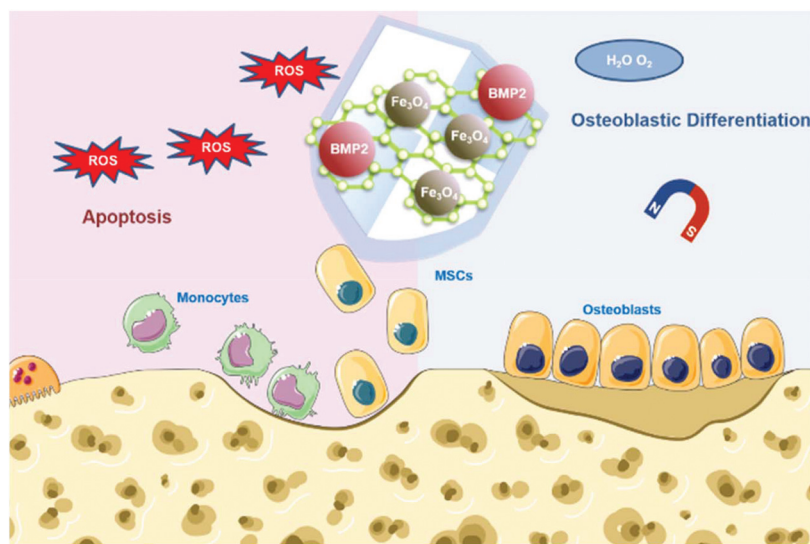
To make full use of these advantages, herein we designed Fe₃O₄@GO magnetic nanocomposites (Fe₃O₄@GO MNCs) to work synergistically in bone tissue engineering. The Fe₃O₄@GO MNCs eliminated active intermediates produced by Fe₃O₄ in the Fenton reaction by capturing $\cdot\text{OH}$, providing a mild scavenging activity to regulate ROS while retaining intrinsic magnetic property. The Fe₃O₄@GO MNCs showed a cytoprotective effect *in vitro*. We further loaded BMP2 growth factor onto Fe₃O₄@GO MNCs, demonstrating the enhanced osteogenic differentiation of MSCs *in vitro* under magnetic regulation. Our results demonstrated an effective strategy for the protection of endogenous/transplanted stem cells and

improvement of the osteogenic effect in tissue engineering by both utilizing the magnetic field and regulating the ROS level (Scheme 1).

Results and discussion

Characterization of Fe₃O₄@GO MNCs

Commercially available GO and Fe₃O₄@GO MNCs were used. Fe₃O₄ NPs were synthesized by a hydrothermal method.⁴⁶ GO, Fe₃O₄, and Fe₃O₄@GO MNCs were first characterized by transmission electron microscopy (TEM) imaging. Fig. 1(a) clearly showed a typical GO structure of sheet like regions with a few wrinkles. As shown in Fig. 1(b), irregular Fe₃O₄ NPs with an average size of 12 nm were observed. For Fe₃O₄@GO MNCs shown in Fig. 1(c), a large amount of Fe₃O₄ NPs were observed on sheet like GO. TEM images confirmed the successful assembly of Fe₃O₄ onto GO to form Fe₃O₄@GO MNCs. The Fourier transform infrared (FTIR) spectrum of GO (Fig. 1(d)) showed a broad adsorption peak around 3500 cm⁻¹, attributed to the stretching vibration of -COOH and -OH, in line with the negative potential of GO shown in Fig. S2.† The absorption peaks at 2800–3000 cm⁻¹ and 1750 cm⁻¹ were characteristic peaks of the alkyl group and C=O, respectively. The peak at 1358 cm⁻¹ was assigned to the C–O–C stretching vibration. The FTIR spectrum of Fe₃O₄@GO MNCs showed absorption peaks at 570, 1738, and 3430 cm⁻¹, which corresponded to the stretching vibration peak of Fe–O, C=O, and O–H, respectively. Powder X-ray diffraction (PXRD) was also used to determine the chemical composition and phase purity of Fe₃O₄@GO MNCs. Fig. 1(e) shows the diffraction peak of GO (black line) at 2θ = 11.7°, and there was a broad diffraction peak at 2θ = 24.8°. Fe₃O₄@GO MNCs (blue line) retained the characteristic diffraction peaks of Fe₃O₄ (red line) at 2θ = 30°, 35°, 43°, 57°, and 63°. The XRD profile of GO exhibited a strong single



Scheme 1 Schematic illustration of Fe₃O₄@GO/BMP2 protecting mesenchymal stem cells and promoting osteogenic differentiation.

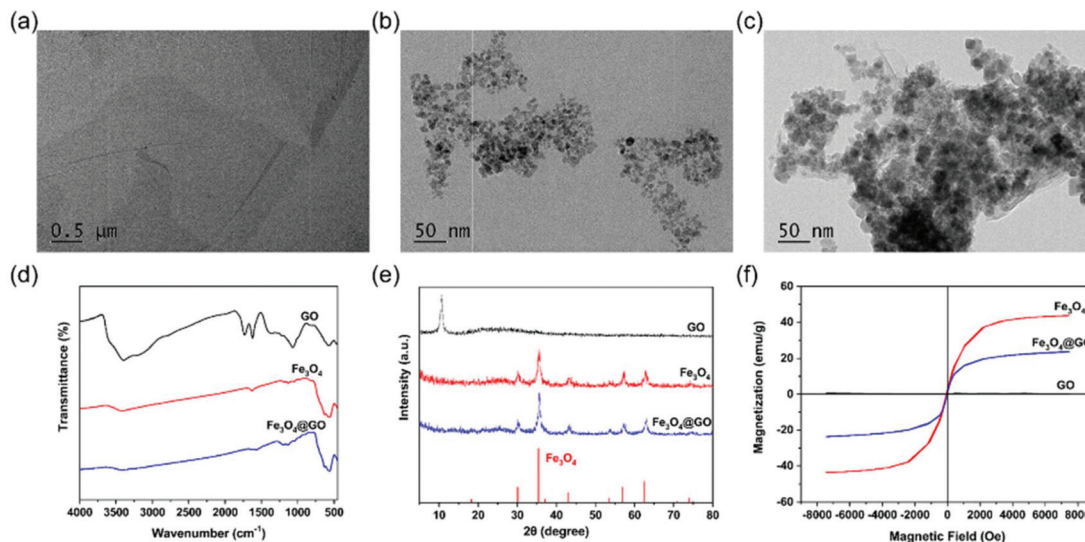


Fig. 1 NP characterization. TEM images of (a) GO, (b) Fe₃O₄, and (c) Fe₃O₄@GO MNCs. (d) FTIR spectra of GO, Fe₃O₄, and Fe₃O₄@GO MNCs. (e) XRD patterns of GO, Fe₃O₄, and Fe₃O₄@GO MNCs (the red lines at the bottom mark the reference pattern of Fe₃O₄ from the JCPDS database, card number 19-0629). (f) Room-temperature hysteresis curves of GO, Fe₃O₄, and Fe₃O₄@GO MNCs.

reflection at $2\theta = 11.7^\circ$, which corresponds to an interlayer d_{001} spacing of 0.8 nm. This spacing is in line with the XRD characteristic values for GO in previous reports.^{47,48} Although it was not obvious to see the XRD peak of GO in the curve of Fe₃O₄@GO MNCs, FTIR and XPS results still clearly demonstrated the presence of GO in Fe₃O₄@GO MNCs. We attributed the missing peak of GO at 11.7° to the great amount difference between Fe₃O₄ and GO, which can be proved in TEM imaging. In Fig. 1c, most of the GO was shielded by Fe₃O₄, thereby weakening the XRD signal of GO. A similar shielding effect of XRD in Fe₃O₄@GO MNCs was also found in a previous report.⁴⁹ The hysteresis loops in Fig. 1(f) show the retained superparamagnetic behavior of Fe₃O₄@GO MNCs (blue line), which was derived from Fe₃O₄ (red line). The saturation magnetization of Fe₃O₄@GO MNCs was about 23 emu g⁻¹, while that of Fe₃O₄ reached about 43 emu g⁻¹.

H₂O₂ and hydroxyl radical scavenging activity of Fe₃O₄@GO MNCs

H₂O₂, a sort of highly oxidative ROS, has been found to be closely related to tissue damage. We used the Ti(SO₄)₂ method to detect the hydrogen peroxide scavenging activity of Fe₃O₄@GO MNCs by monitoring the colorimetric signal at 405 nm, which originated from the yellowish oxidized product of Ti(SO₄)₂ in the presence of H₂O₂. To mimic the mild environment in the organism, this reaction was carried out in a buffer at pH 7.4. As shown in Fig. 2(a), both Fe₃O₄@GO MNCs (blue line) and Fe₃O₄ (red line) exhibited H₂O₂ scavenging activity within 3 h. Notably, Fe₃O₄@GO MNCs scavenged more than 30% of H₂O₂ within 3 h while Fe₃O₄ scavenged less than 15% of H₂O₂. This result showed the much higher H₂O₂ scavenging activity of Fe₃O₄@GO MNCs than Fe₃O₄ at pH 7.4. As a comparison, GO showed negligible H₂O₂ scavenging

activity. Moreover, to better mimic the acidic environment of inflammation, the H₂O₂ scavenging activity of the three materials was also investigated at pH 5.8 and a similar trend was obtained (Fig. S3†). Strikingly, as shown in Fig. 2(b), Fe₃O₄@GO MNCs presented better activity on degrading H₂O₂ at pH 7.4 (red line).

$\cdot\text{OH}$ is a stronger ROS which could be generated from the Fe₃O₄-mediated Fenton reaction. Effective scavenging/capturing of $\cdot\text{OH}$ is beneficial for protecting cells and growth factors from ROS-induced damage. To characterize the $\cdot\text{OH}$ scavenging ability of Fe₃O₄@GO MNCs, electron paramagnetic resonance spectroscopy (EPR) was applied with a capture agent 5,5-dimethyl-1-pyridine N-oxide (DMPO) which can specifically capture $\cdot\text{OH}$, and generate resultant DMPO/ $\cdot\text{OH}$ with distinct characteristic lines of 1 : 2 : 2 : 1. The strength of the characteristic peak was used for quantitatively monitoring $\cdot\text{OH}$ levels. As shown in Fig. 2(c), the reaction of Fe₃O₄ with H₂O₂ produced a large amount of $\cdot\text{OH}$ radicals (red line). In contrast, the reaction of Fe₃O₄@GO MNCs (blue line) or GO (black line) with H₂O₂ produced very similar peak signals to the control (H₂O₂ alone, green line), indicating that almost no $\cdot\text{OH}$ was produced in the reactions. This result demonstrated that the ROS regulation ability of Fe₃O₄@GO MNCs could eliminate the $\cdot\text{OH}$ radicals *in situ* produced by Fe₃O₄. To further confirm the $\cdot\text{OH}$ radical scavenging ability of Fe₃O₄@GO MNCs, FeSO₄ and H₂O₂ (1 : 1) were mixed with NPs and the level of $\cdot\text{OH}$ radicals was detected by measurement of EPR. Interestingly, there was no significant difference in the peak intensity of DMPO/ $\cdot\text{OH}$ for GO (Fig. 2(e)) and Fe₃O₄ (Fig. 2(f)) groups, compared to the control group (Fig. 2(d)), while Fe₃O₄@GO MNCs (Fig. 2(g) and (h)) obviously reduced $\cdot\text{OH}$ radicals. The above results indicated that Fe₃O₄@GO MNCs not only scavenged the *in situ* generated $\cdot\text{OH}$, but also scavenged $\cdot\text{OH}$ radicals in the environ-

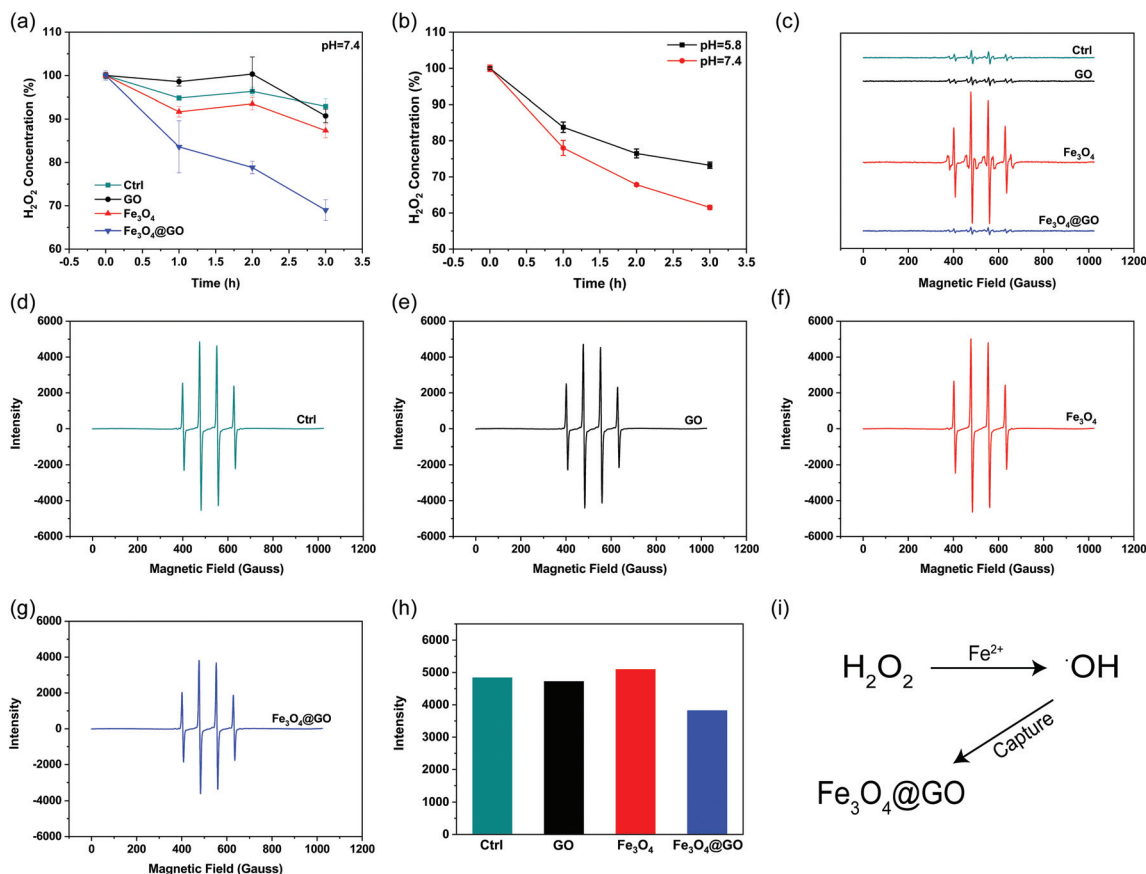


Fig. 2 H₂O₂ and hydroxyl radical scavenging activity of NPs. (a) H₂O₂ scavenging activity of indicated NPs (pH 7.4). (b) H₂O₂ scavenging activity of Fe₃O₄@GO MNCs in pH 7.4 and pH 5.8. (c) Hydroxyl radical scavenging activity. EPR spectra of samples containing DMPO, Fe²⁺, and H₂O₂ in the absence and presence of indicated NPs. (d–g) EPR spectra of samples containing DMPO, FeSO₄, and H₂O₂ in the absence (d) and presence of (e) GO, (f) Fe₃O₄, and (g) Fe₃O₄@GO MNCs, respectively. EPR spectrum of DMPO only was collected for comparison. (h) Content of ·OH radicals indicated by EPR intensity in (d–g). (i) Schematic diagram of Fe₃O₄@GO MNCs capturing hydroxyl radicals in the Fenton reaction. Each error bar shows the standard deviation of three independent measurements.

ment. The above results indicated that Fe₃O₄@GO MNCs do not produce ·OH even in the presence of H₂O₂ in a biological environment. Although Fe₃O₄@GO MNCs did not exhibit high SOD-like activity (Fig. S4†), they regulated ROS through their H₂O₂ scavenging activity and ·OH scavenging activity.

Intracellular ROS scavenging

Based on the enzyme-like activity of Fe₃O₄@GO MNCs, we attempted to apply the nanomaterial to biological environments. The cytotoxicity of Fe₃O₄@GO MNCs was first evaluated by using the Cell Counting Kit-8 (CCK-8) assay. As shown in Fig. 3(a), MSCs steadily proliferated when incubated with different concentrations of Fe₃O₄@GO MNCs, and there were no significant differences among 10 and 25 μg mL⁻¹ at each time point. The CCK-8 assay demonstrated the good biocompatibility of Fe₃O₄@GO MNCs.

After confirmation of the good biocompatibility of Fe₃O₄@GO MNCs, we further explored whether Fe₃O₄@GO MNCs could protect the transplanted MSCs from ROS-induced apoptosis. To mimic ROS in a biological environment, we used

different concentrations of H₂O₂ to induce apoptosis and analyzed the proportion of apoptosis by using flow cytometry (Fig. S5†). A sublethal dose of H₂O₂ (100 μM) was selected for following research. Furthermore, we incubated NPs with 100 μM H₂O₂ pre-treated MSCs. According to the results shown in Fig. 3(b) and (c), Fe₃O₄@GO MNCs reduced apoptosis caused by H₂O₂, showing protection on MSCs (Fig. S6†). Interesting, a low concentration of Fe₃O₄@GO MNCs (10 μg mL⁻¹) even showed better cytoprotection than a high concentration (25 μg mL⁻¹). However, there was no protection effect in either Fe₃O₄ or GO (10 μg mL⁻¹ and 25 μg mL⁻¹).

Considering that complex tissue regeneration environments are often accompanied by inflammation, a LPS induced inflammation model was established to investigate whether ROS regulated Fe₃O₄@GO MNCs have anti-inflammation effect. Again, a sublethal concentration of LPS was used to stimulate MSCs. Then 2',7'-dichlorofluorescein diacetate (DCFH-DA) was used as a fluorescent probe to monitor the intracellular ROS level. Fluorescence images (Fig. 3(d) and (e)) confirmed the increased ROS level in the LPS-treated group,

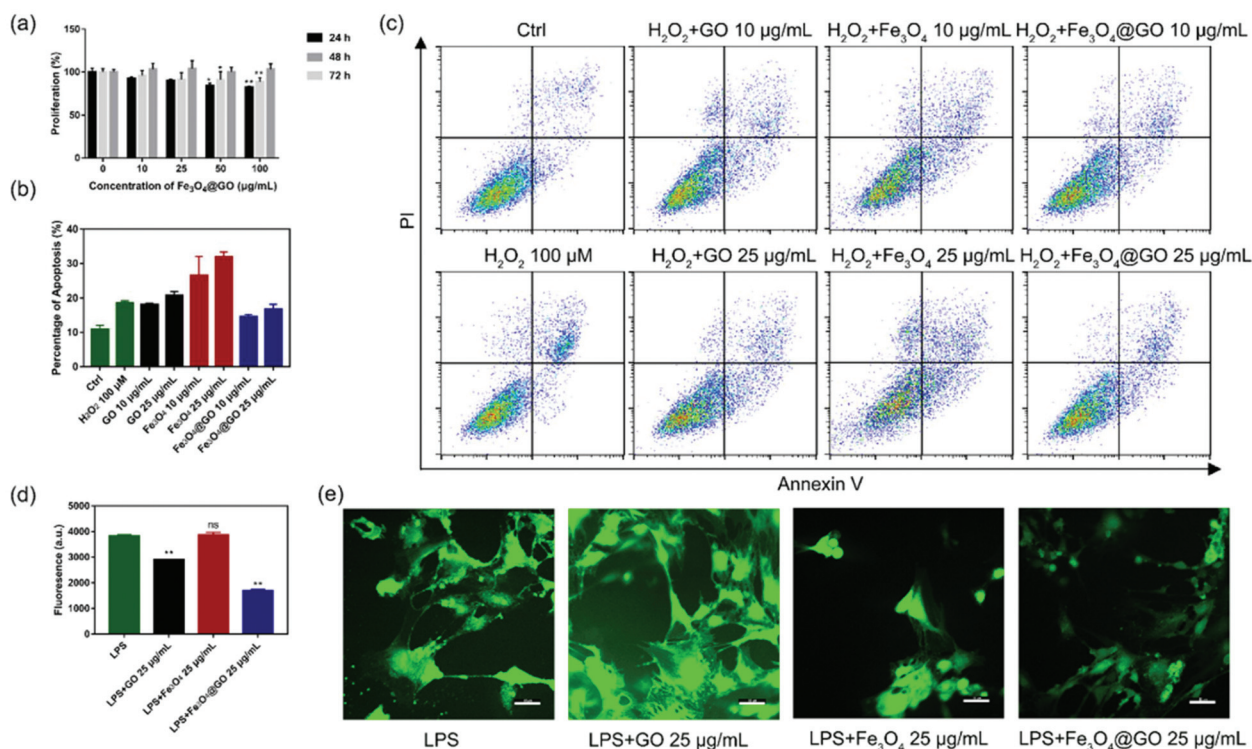


Fig. 3 (a) Cell viability of MSCs incubated with different concentrations of Fe₃O₄@GO MNCs at different times. (b) Apoptosis level of MSCs cultured with 100 µM H₂O₂ after incubation with different concentrations of NPs. (c) Corresponding flow cytometry plots. (d) DCF fluorescence values of MSCs treated with LPS following pre-treatment with indicated NPs (e). The intracellular ROS levels were indicated by the green fluorescence of DCF. Scale bar: 50 µm. Each error bar shows the standard deviation of three independent measurements.

excluding the possibility that nanomaterials induced ROS (Fig. S8[†]). What's more, Fe₃O₄@GO MNCs effectively reduced ROS with lower fluorescence intensity. However, the Fe₃O₄ group showed an enhancement of fluorescence and changes in cell morphology. These results revealed the ability of Fe₃O₄@GO MNCs to protect MSCs by scavenging intercellular ROS, which ultimately led to higher survival and therapeutic efficacy of the transplanted MSCs.

Fe₃O₄@GO MNCs induced osteogenic differentiation of MSCs

In order to create an osteogenic favored artificial environment under complicated pathological conditions, we not only utilized the magnetic responsiveness from Fe₃O₄, but also took advantage of the delivery ability from GO. First, BMP2, a classic growth factor with excellent osteogenic induction capacity, was loaded on Fe₃O₄@GO to fabricate Fe₃O₄@GO/BMP2 by incubating Fe₃O₄@GO MNCs with BMP2 solution for 24 h. Subsequently, we evaluated the amount of BMP2 loaded onto the NPs by a centrifugation analysis. After centrifugation, the residual BMP2 in the supernatant was stained with Coomassie Brilliant Blue G250. The concentrations of BMP2 loaded by NPs used for MSC cultures can then be calculated with reference to the standard curve. As shown in Fig. S9[†], Fe₃O₄@GO MNCs showed higher BMP2 loading than Fe₃O₄ NPs, where 330 ng mL⁻¹, 250 ng mL⁻¹, and 167 ng mL⁻¹ of BMP2 were loaded by 25 µg mL⁻¹ of Fe₃O₄, GO, and

Fe₃O₄@GO, respectively. The BMP2 loading capability of NPs was well correlated with the zeta potential results, where the more negatively charged NPs had a higher loading capacity of BMP2 (Fig. S2[†]). Next, we co-cultured MSCs with NPs/BMP2. The RNA level of osteogenic markers was detected 7 d later. It can be found that Fe₃O₄@GO/BMP2 had a good advantage in promoting osteogenic differentiation of MSC cells, no matter at a low concentration (10 µg mL⁻¹) (Fig. 4(a)) or at a high concentration (25 µg mL⁻¹, Fig. 4(b)). Furthermore, to detect the effects of magnetic field response from NPs on osteogenic induction, we co-cultured MSCs with Fe₃O₄@GO/BMP2 and GO/BMP2 under a magnetic field. Apparently, Fe₃O₄@GO/BMP2 with magnetic responsiveness had superior osteogenic ability to GO/BMP2 without magnetic responsiveness (Fig. 4(c) and (d)). This advantage was more prominent at the lower concentration (10 µg mL⁻¹) (Fig. 4(c)). We further compared the osteogenic induction results of two concentrations of Fe₃O₄@GO/BMP2 (Fig. 4(e) and (f)). RT-PCR results showed that both of the two concentrations showed good osteogenic capacity. The results of western blot (Fig. 4(g) and (h)) are consistent with the above results. Besides, staining alkaline phosphatase (ALP) (Fig. 5) showed that Fe₃O₄@GO/BMP2 exhibited better osteogenic ability than GO/BMP2 and Fe₃O₄/BMP2. Moreover, this advantage of Fe₃O₄@GO/BMP2 became more pronounced with the use of SMF, and the alizarin red staining and quantification of calcium nodule formation (Fig. 6) as a

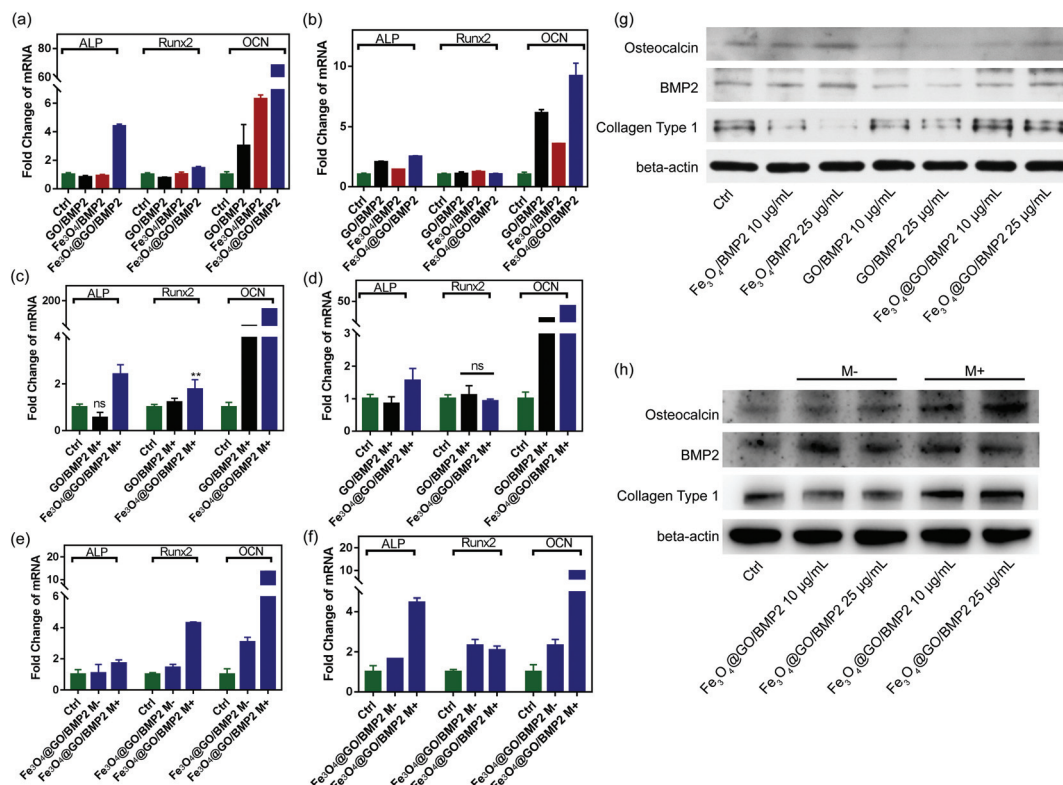


Fig. 4 qPCR analysis of ALP, Runx2, and OCN mRNA expression of MSCs *in vitro* cultured with different concentrations ((a) 10 $\mu\text{g mL}^{-1}$ and (b) 25 $\mu\text{g mL}^{-1}$, respectively) of GO/BMP2, $\text{Fe}_3\text{O}_4/\text{BMP2}$, and $\text{Fe}_3\text{O}_4@\text{GO}/\text{BMP2}$ on day 7. qPCR analysis of ALP, Runx2, and OCN mRNA expression of MSCs *in vitro* cultured with different concentrations ((c) 10 $\mu\text{g mL}^{-1}$ and (d) 25 $\mu\text{g mL}^{-1}$, respectively) of GO/BMP2 and $\text{Fe}_3\text{O}_4@\text{GO}/\text{BMP2}$ in the presence of a static magnetic field on day 7. qPCR analysis of ALP, Runx2, and OCN mRNA expression of MSCs *in vitro* cultured with different concentrations ((e) 10 $\mu\text{g mL}^{-1}$ and (f) 25 $\mu\text{g mL}^{-1}$, respectively) of $\text{Fe}_3\text{O}_4@\text{GO}/\text{BMP2}$ in the absence and presence of a static magnetic field on day 7. (g) Western blotting results of MSCs cultured with different concentrations of GO/BMP2, $\text{Fe}_3\text{O}_4/\text{BMP2}$, and $\text{Fe}_3\text{O}_4@\text{GO}/\text{BMP2}$ on day 7. (h) Western blotting results of MSCs cultured with different concentrations of $\text{Fe}_3\text{O}_4@\text{GO}/\text{BMP2}$ in the absence and presence of a static magnetic field on day 7. Each error bar shows the standard deviation of three independent measurements (* $p < 0.05$, ** $p < 0.01$, ns = no significance).

late osteogenic signal also demonstrated that $\text{Fe}_3\text{O}_4@\text{GO}/\text{BMP2}$ possessed better osteogenic ability with SMF.

Experimental

Chemicals

GO was provided by Hangzhou Gaoxi Technology Co., Ltd. $\text{Fe}_3\text{O}_4@\text{GO}$ MNCs was provided by Nanjing XFNANO Materials Tech. Co., Ltd. Titanium sulfate ($\text{Ti}(\text{SO}_4)_2$) was purchased from Shanghai Macklin Biochemical Co., Ltd. Hydrogen peroxide (H_2O_2 , 30%) and dihydroethidine were purchased from Aladdin Chemical Reagent Co., Ltd. Xanthine and xanthine oxidase were obtained from Sigma-Aldrich. 5,5-Dimethyl-1-pyridine N-oxide (DMPO) was purchased from Nanjing Tongquan Chemical Reagent Co., Ltd. 2',7'-Dichlorofluorescein diacetate (DCFH-DA) and BCIP/NBT Alkaline Phosphatase Color Development Kit were purchased from Beyotime Chemical Reagent Co., Ltd. Cell Counting Kit-8 (CCK-8) was purchased from KeyGen Biotech Co., Ltd. Annexin V-FITC/PI Apoptosis Detection Kit and SYBR Green PCR Master Mix were purchased from Vazyme Biotech Co., Ltd. Trizol reagent was purchased

from Invitrogen Life Technology Co., Ltd. All chemical reagents were used as received without further purification. All aqueous solutions were prepared with deionized water (18.2 $\text{M}\Omega$ cm, Millipore).

Instrumentation

Transmission electron microscopy (TEM) (Tecnai F20) was used at an acceleration voltage of 200 kV. Powder X-ray diffraction (PXRD) data were obtained on a Rigaku Ultima diffractometer by using $\text{Cu K}\alpha$ radiation. X-ray photoelectron spectroscopy (XPS) was performed using a PHI 5000 VersaProbe (UIVAC-PHI, Japan). Electron paramagnetic resonance (EPR) spectra were recorded by using an EMX-10/12 EPR spectrometer (Bruker, Germany). Fluorescence microscopy images were captured by using a confocal fluorescence microscope (CFM, A1, Nikon, Japan).

Preparation of Fe_3O_4 NPs

Fe_3O_4 NPs were prepared following a reported method.⁴⁶ FeCl_2 (0.2 M, 1.0 mL) and FeCl_3 solutions (0.1 M, 4.0 mL) were first mixed under nitrogen gas. Aqueous ammonia (0.2 M, 15 mL)

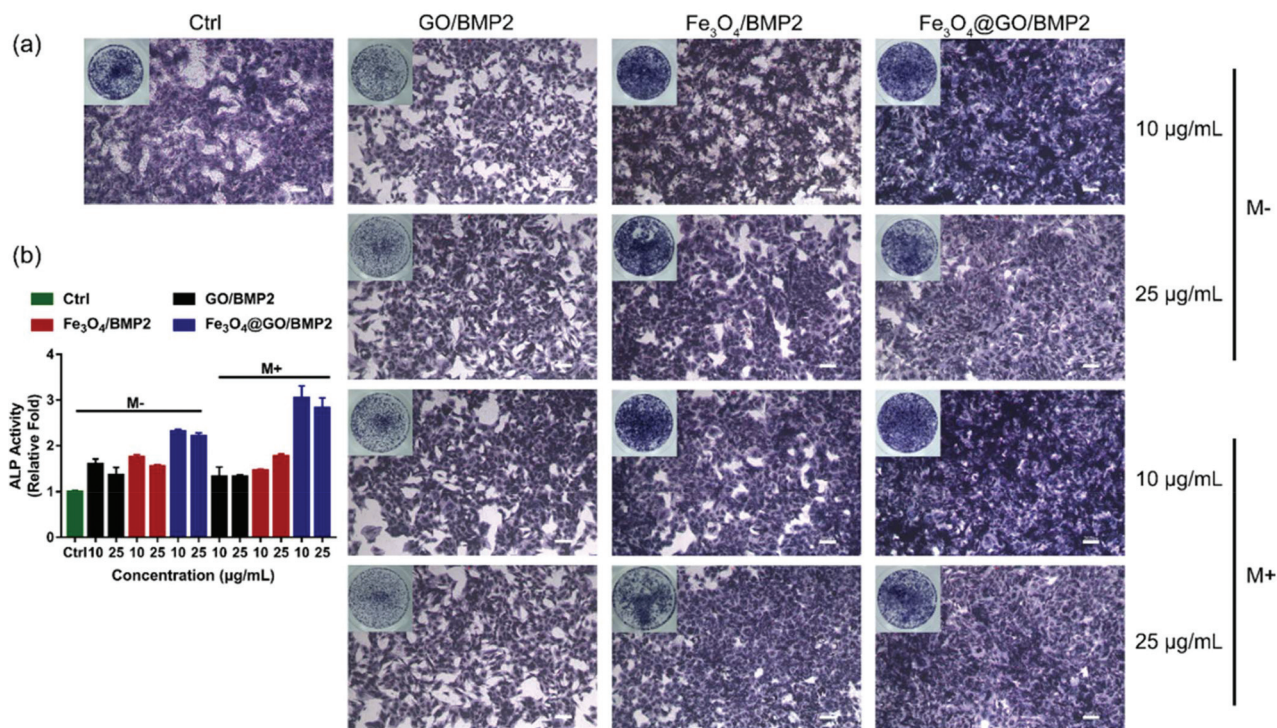


Fig. 5 (a) ALP staining and (b) relative ALP activities of MSCs *in vitro* cultured with different concentrations of NPs/BMP2 in the absence (M-) and presence (M+) of a static magnetic field on day 7. Scale bar: 200 µm. Each error bar shows the standard deviation of three independent measurements.

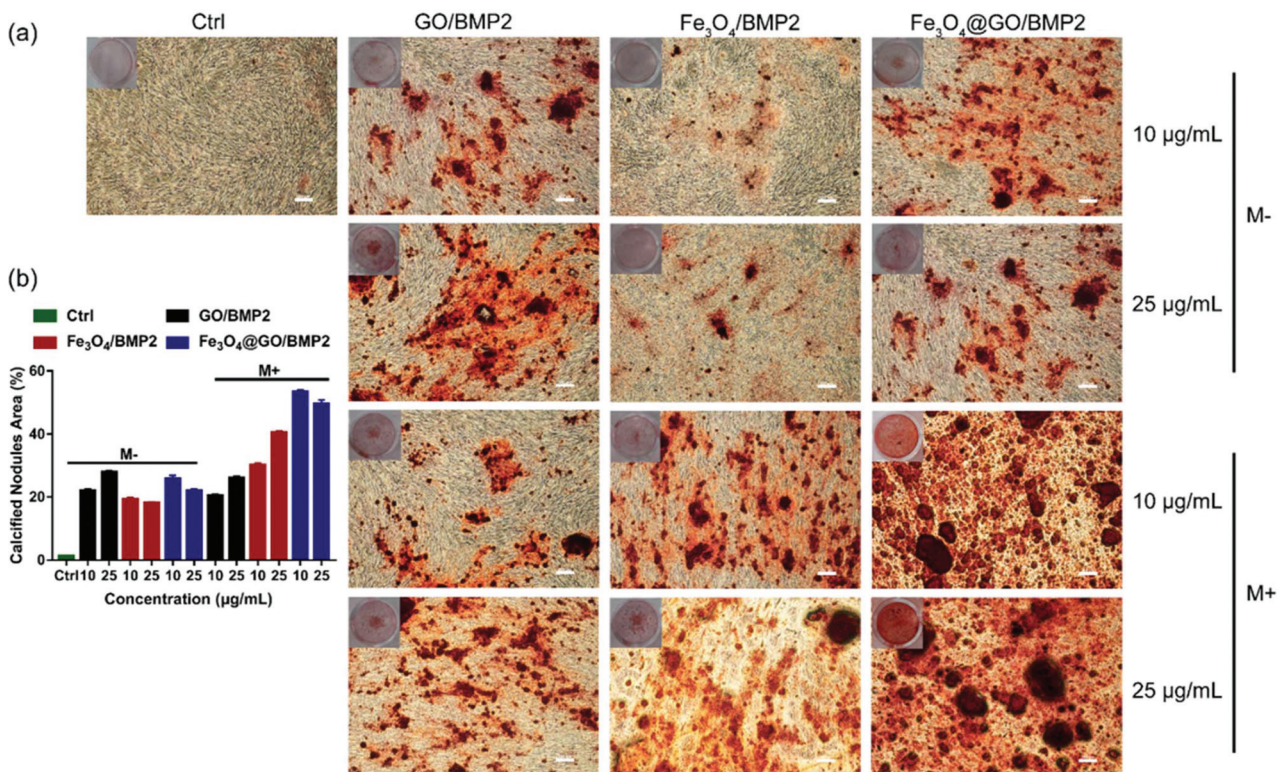


Fig. 6 (a) Alizarin red staining and (b) quantification of calcium nodule formation of MSCs *in vitro* cultured with different concentrations of NPs/BMP2 in the absence (M-) and presence (M+) of a static magnetic field on day 21. Scale bar: 200 µm. Each error bar shows the standard deviation of three independent measurements.

was added dropwise to the mixture under stirring. Then the mixture was heated under nitrogen at 80 °C for 1 h. After cooling to room temperature, the obtained Fe₃O₄ NPs were centrifuged and washed with Milli-Q water.

Hydrogen peroxide scavenging activity of NPs

The hydrogen peroxide scavenging activity of NPs was studied by measuring the consumption of H₂O₂ by the Ti(SO₄)₂ colorimetric method.⁵⁰ First, NPs of different concentrations (10 µg mL⁻¹ and 25 µg mL⁻¹) were mixed with 5 mL of H₂O₂ (1 × 10⁻³ M) and then incubated together at 37 °C with shaking, and marked as solution A. 1.33 mL of Ti(SO₄)₂ (24%) and 8.33 mL of H₂SO₄ were added to water to obtain 50 mL of solution B. Then 100 µL of solution A was mixed with 200 µL of solution B thoroughly. After 10 min, NPs were removed by centrifugation, and the absorbance at 405 nm of the supernatant was measured using a UV-Vis spectrophotometer to detect the remaining H₂O₂. Time dependent hydrogen peroxide scavenging activity was recorded as described every 1 h.

Hydroxyl radical scavenging activity of NPs

The ·OH was detected by using EPR, where the DMPO/·OH spin adduct was produced and had characteristic lines with relative intensities of 1:2:2:1. The NPs were added into Fenton agents of 0.2 mM FeSO₄ and 0.2 mM H₂O₂ for 2 min. Subsequently, samples were detected by EPR.

Measurement of SOD-like activity

Xanthine and xanthine oxidase were reacted in phosphate buffer (0.1 M, pH 7.4, 37 °C) for 1 h to produce ·O₂⁻.⁵¹ At the same time, a certain amount of nanomaterials was added to eliminate the generated ·O₂⁻. Forty minutes later, 0.5 mg mL⁻¹ dihydroethidine (HE) in dimethyl sulfoxide was added. After further incubation for 40 min at 37 °C, the fluorescence of the final mixture was measured.

Cytotoxicity assay

Cytotoxicity of nanocomposites was tested by using the Cell Counting Kit-8 assay. MSCs (3 × 10³ per well) were cultured in 96-well plates at 37 °C and 5% CO₂. Then, 10 µL of CCK-8 solution was added to the medium of cells. The absorbance at 450 nm was measured after incubation for 2 h.

Flow cytometry analysis of apoptosis

Treated MSCs were collected and washed with PBS, then resuspended in binding buffer. Next, MSCs were incubated with fluorescein-labelled Annexin V and propidium iodide (PI) prior to flow cytometry. An FITC signal detector (FL1, 525 nm) and a detector reserved for phycoerythrin emission (FL2, 575 nm) were used to determine two kinds of signals. Analysis of data was performed by using CellQuest Software from BD.

PCR analysis

Total RNA was extracted by using Trizol reagent. Then RNA was reversely transcribed to cDNA by using a PrimeScript™ II First-Strand Synthesis System. qPCR was performed by using SYBR

Table 1 Primer sequences used in qPCR analysis

Gene	Primer sequence
β-Actin	Forward 5'-GGAGATTACTGCCCTGGCTCCTA-3' Reverse 5'-GACTCATCGTACTCCTGCTTGCTG-3'
ALP	Forward 5'-CATCGCCTATCAGCTAATGCACA-3' Reverse 5'-ATGAGGTCCAGGCCATCCAG-3'
Runx2	Forward 5'-CATGGCCGGGAATGATGAG-3' Reverse 5'-TGTGAAGACCGTTATGGTCAAAGTG-3'
OCN	Forward 5'-ATGAGGACCTCTCTCTGCTC-3' Reverse 5'-CTAAACGGTGGTGCCATAGAT-3'

Green PCR Master Mix, and the resulting Ct values were normalized to GAPDH. All primer sequences are provided in Table 1.

Statistical analysis

Statistical analyses were performed by using GraphPad Prism 7.0. Data were represented as the mean ± standard deviation. One-way analysis of variance (ANOVA) was performed to evaluate the effect of NPs. *p* values <0.05 were considered statistically significant.

Conclusions

In conclusion, ROS scavenging Fe₃O₄@GO MNCs were developed. GO effectively reduced the hydroxyl radicals which were produced by the Fe₃O₄-mediated Fenton reaction, and thus resulted in moderate scavenging of ROS. Because of the ROS scavenging and magnetic properties, the Fe₃O₄@GO MNCs protected the MSCs from ROS damage. When delivered by using Fe₃O₄@GO MNCs, BMP2 functioned in the inflammatory and oxidative stress environment involved in tissue damage. In addition, SMF as a non-invasive stimulation enhanced osteogenic ability *in vitro*. This work not only developed a method to fabricate multifunctional nanocomposites, but also proved the delicate regulation of ROS level by Fe₃O₄@GO MNCs, a prerequisite in the field of tissue regeneration, especially for stem cell implantation and growth factor delivery.

Conflicts of interest

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript. There are no conflicts to declare.

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Notes and references

- S. D. Subramony, B. R. Dargis, M. Castillo, E. U. Azeloglu, M. S. Tracey, A. Su and H. H. Lu, *Biomaterials*, 2013, **34**, 1942–1953.
- M. De Mattei, A. Caruso, G. C. Traina, F. Pezzetti, T. Baroni and V. Sollazzo, *Bioelectromagnetics*, 1999, **20**, 177–182.
- K. J. McLeod and L. Collazo, *Radiat. Res.*, 2000, **153**, 706–714.
- T. A. Ulrich, E. M. de Juan Pardo and S. Kumar, *Cancer Res.*, 2009, **69**, 4167–4174.
- G. Altman, R. Horan, I. Martin, J. Farhadi, P. Stark, V. Volloch, J. Richmond, G. Vunjak-Novakovic and D. L. Kaplan, *FASEB J.*, 2002, **16**, 270–272.
- H. Huang, S. Delikanli, H. Zeng, D. M. Ferkey and A. Pralle, *Nat. Nanotechnol.*, 2010, **5**, 602–606.
- C. Shuai, L. Yu, P. Feng, Y. Zhong, Z. Zhao, Z. Chen and W. Yang, *Mater. Chem. Front.*, 2020, **4**, 2398–2408.
- J. H. Noar and R. D. Evans, *Br. J. Orthod.*, 1999, **26**, 29–37.
- D. H. Trock, *Rheum. Dis. Clin. North Am.*, 2000, **26**, 51–62.
- B. Thiesen and A. Jordan, *Int. J. Hyperthermia*, 2008, **24**, 467–474.
- H. Kotani, H. Kawaguchi, T. Shimoaka, M. Iwasaka, S. Ueno, H. Ozawa, K. Nakamura and K. Hoshi, *J. Bone Miner. Res.*, 2002, **17**, 1814–1821.
- C. A. L. Bassett, S. N. Mitchell and S. R. Gaston, *J. Am. Med. Assoc.*, 1982, **247**, 623–628.
- S. Hao, J. Meng, Y. Zhang, J. Liu, X. Nie, F. Wu, Y. Yang, C. Wang, N. Gu and H. Xu, *Biomaterials*, 2017, **140**, 16–25.
- C. Shuai, W. Yang, C. He, S. Peng, C. Gao, Y. Yang, F. Qi and P. Feng, *Mater. Des.*, 2020, **185**, 108275.
- W. Zhang, G. Yang, X. Wang, L. Jiang, F. Jiang, G. Li, Z. Zhang and X. Jiang, *Adv. Mater.*, 2017, **29**, 1703795.
- Y. Yu, S. Ren, Y. Yao, H. Zhang, C. Liu, J. Yang, W. Yang and L. Miao, *J. Biomed. Nanotechnol.*, 2017, **13**, 835–847.
- H. Isomura, K. Fujie, K. Shibata, N. Inoue, T. Iizuka, G. Takebe, K. Takahashi, J. Nishihira, H. Izumi and W. Sakamoto, *Toxicology*, 2004, **197**, 92–99.
- Y. Hamada, H. Fujii and M. Fukagawa, *Bone*, 2009, **45**, S35–S38.
- S. Leutner, A. Eckert and W. Müller, *J. Neural Transm.*, 2001, **108**, 955–967.
- E. Cabiscol, J. Tamarit and J. Ros, *Int. Microbiol.*, 2000, **3**(1), 3–8.
- R. A. Floyd and K. Hensley, *Neurobiol. Aging*, 2002, **23**, 795–807.
- K. Ito, A. Hirao, F. Arai, K. Takubo, S. Matsuoka, K. Miyamoto, M. Ohmura, K. Naka, K. Hosokawa and Y. Ikeda, *Nat. Med.*, 2006, **12**, 446–451.
- E. R. Stadtman, *Free Radical Biol. Med.*, 1990, **9**, 315–325.
- D. Brown, K. Donaldson, P. Borm, R. Schins, M. Dehnhardt, P. Gilmour, L. Jimenez and V. Stone, *Am. J. Physiol.: Lung Cell. Mol. Physiol.*, 2004, **286**, L344–L353.
- S. Basu, K. Michaëlsson, H. Olofsson, S. Johansson and H. Melhus, *Biochem. Biophys. Res. Commun.*, 2001, **288**, 275–279.
- S. Reuter, S. C. Gupta, M. M. Chaturvedi and B. B. Aggarwal, *Free Radical Biol. Med.*, 2010, **49**, 1603–1616.
- I. L. Chapple and J. B. Matthews, *Periodontology 2000*, 2007, **43**, 160–232.
- P. Jia, Y. J. Xu, Z. L. Zhang, K. Li, B. Li, W. Zhang and H. Yang, *J. Orthop. Res.*, 2012, **30**, 1843–1852.
- J. Hoffmann, A. J. Glassford, T. C. Doyle, R. C. Robbins, S. Schrepfer and M. P. Pelletier, *Thorac. Cardiovasc. Surg.*, 2010, **58**, 136–142.
- P. K. Nguyen, J. Riegler and J. C. Wu, *Cell Stem Cell*, 2014, **14**, 431–444.
- F. Atashi, A. Modarressi and M. S. Pepper, *Stem Cells Dev.*, 2015, **24**, 1150–1163.
- F. Wauquier, L. Leotoing, V. Coxam, J. Guicheux and Y. Wittrant, *Trends Mol. Med.*, 2009, **15**, 468–477.
- I. J. M. De Vries, W. J. Lesterhuis, J. O. Barentsz, P. Verdijk, J. H. Van Krieken, O. C. Boerman, W. J. Oyen, J. J. Bonenkamp, J. B. Boezeman, G. J. Adema, J. W. Bulte, T. W. Scheenen, C. J. Punt, A. Heerschap and C. G. Figdor, *Nat. Biotechnol.*, 2005, **23**, 1407–1413.
- D. Liu, W. Wu, J. Ling, S. Wen, N. Gu and X. Zhang, *Adv. Funct. Mater.*, 2011, **21**, 1498–1504.
- L. Gao, J. Zhuang, L. Nie, J. Zhang, Y. Zhang, N. Gu, T. Wang, J. Feng, D. Yang, S. Perrett and X. Yan, *Nat. Nanotechnol.*, 2007, **2**, 577–583.
- H. Wei and E. Wang, *Anal. Chem.*, 2008, **80**, 2250–2254.
- H. Wei and E. Wang, *Chem. Soc. Rev.*, 2013, **42**, 6060–6093.
- N. Singh, G. J. Jenkins, R. Asadi and S. H. Doak, *Nano Rev.*, 2010, **1**, 5358.
- J. A. Imlay, S. M. Chin and S. Linn, *Science*, 1988, **240**, 640–642.
- B. Yang, Y. Chen and J. Shi, *Chem. Rev.*, 2019, **119**, 4881–4985.
- Y. Song, K. Qu, C. Zhao, J. Ren and X. Qu, *Adv. Mater.*, 2010, **22**, 2206–2210.
- Y. Qiu, Z. Wang, A. C. Owens, I. Kulaots, Y. Chen, A. B. Kane and R. H. Hurt, *Nanoscale*, 2014, **6**, 11744–11755.
- J. Park, B. Kim, J. Han, J. Oh, S. Park, S. Ryu, S. Jung, J.-Y. Shin, B. S. Lee and B. H. Hong, *ACS Nano*, 2015, **9**, 4987–4999.
- W. G. La, S. Park, H. H. Yoon, G. J. Jeong, T. J. Lee, S. H. Bhang, J. Y. Han, K. Char and B. S. Kim, *Small*, 2013, **9**, 4051–4060.
- L. Feng, L. Wu and X. Qu, *Adv. Mater.*, 2013, **25**, 168–186.
- Z. Zhang, X. Zhang, B. Liu and J. Liu, *J. Am. Chem. Soc.*, 2017, **139**, 5412–5419.

- 47 X. Qi, K. Y. Pu, X. Zhou, H. Li, B. Liu, F. Boey, W. Huang and H. Zhang, *Small*, 2010, **6**, 663–669.
- 48 Y. Guo, X. Sun, Y. Liu, W. Wang, H. Qiu and J. Gao, *Carbon*, 2012, **50**, 2513–2523.
- 49 W. Zhang, G. Yang, X. Wang, L. Jiang, F. Jiang, G. Li, Z. Zhang and X. Jiang, *Adv. Mater.*, 2017, **29**, 1703795.
- 50 J. Kim, H. R. Cho, H. Jeon, D. Kim, C. Song, N. Lee, S. H. Choi and T. Hyeon, *J. Am. Chem. Soc.*, 2017, **139**, 10992–10995.
- 51 J. J. Hu, N.-K. Wong, S. Ye, X. Chen, M.-Y. Lu, A. Q. Zhao, Y. Guo, A. C.-H. Ma, A. Y.-H. Leung and J. Shen, *J. Am. Chem. Soc.*, 2015, **137**, 6837–6843.