


 Cite this: *Chem. Commun.*, 2022, 58, 5753

 Received 27th February 2022,
 Accepted 13th April 2022

DOI: 10.1039/d2cc01197b

rsc.li/chemcomm

High capacity ammonia adsorption in a robust metal–organic framework mediated by reversible host–guest interactions†

 Lixia Guo,^a Xue Han,^a Yujie Ma,^a Jiangnan Li,^a Wanpeng Lu,^a Weiyao Li,^a Daniel Lee,^b Ivan da Silva,^c Yongqiang Cheng,^d Svemir Rudić,^c Pascal Manuel,^c Mark D. Frogley,^e Anibal J. Ramirez-Cuesta,^d Martin Schröder^{id}*^a and Sihai Yang^{id}*^a

To understand the exceptional adsorption of ammonia (NH₃) in MFM-300(Sc) (19.5 mmol g⁻¹ at 273 K and 1 bar without hysteresis), we report a systematic investigation of the mechanism of adsorption by a combination of *in situ* neutron powder diffraction, inelastic neutron scattering, synchrotron infrared microspectroscopy, and solid-state ⁴⁵Sc NMR spectroscopy. These complementary techniques reveal the formation of reversible host–guest supramolecular interactions, which explains directly the observed excellent reversibility of this material over 90 adsorption–desorption cycles.

Annual global production of ammonia (NH₃) is around 170 million tonnes reflecting its role as a major feedstock for agriculture and industry.¹ The high hydrogen content (17.8 wt %) and hydrogen volume density (105 kg m⁻³) of NH₃ make it a desirable carbon-free hydrogen carrier, and NH₃ is therefore regarded as a surrogate for the H₂ economy. However, the corrosive and toxic nature of NH₃ makes the development of stable storage materials with high and reversible uptakes extremely challenging. Conventional sorbent materials such as zeolites,² activated carbons,³ and organic polymers⁴ have been investigated for the storage of NH₃, but show low and often irreversible uptakes. Metal–organic framework (MOF) materials have been postulated as promising candidates for gas storage due to their high surface areas and versatile pore structures.⁵ As opposed to conventional adsorbents, the affinities of MOF

materials to a target gas can be tailored by grafting the pore interior with functional groups to anchor the gas through coordination, hydrogen bonding, electrostatic interactions, acid–base interactions or π–π stacking.^{5–7} A large number of MOFs with functional groups (*e.g.* –COOH,⁸ –OH⁹) and open metal sites¹⁰ have been reported to impart enhanced affinity to gas molecules. Several state-of-the-art MOFs, such as MOF-177,¹¹ M₂Cl₂Bbta [Bbta = 1*H*,5*H*-benzo(1,2-*d*:4,5-*d'*)bistriazole; M = Co, Mn],¹² M₂Cl₂(BTDD) {BTDD = bis(1*H*-1,2,3-triazolo[4,5-*b*],[4',5'-*i*])dibenzo[1,4]dioxin; M = Mn, Co, Ni and Cu}¹³ as well as MFM-300(M) (M = Al, Fe, V, Cr, In),^{6,14} have been investigated for NH₃ adsorption. However, due to the reactive and corrosive nature of NH₃, many MOF systems showed structural degradation and/or significant loss of uptake after consecutive cycles owing to irreversible host–guest binding. So far, only a very limited number of MOFs exhibit reversible NH₃ sorption over multiple cycles.^{6,8,13–16} Unravelling the molecular details of host–guest interactions is of critical importance if new efficient NH₃ storage systems are to be developed. This is however highly challenging, not least because hydrogen atoms are invisible in X-ray diffraction experiments and NH₃ molecules can act as a rapid rotor even in solid state.

The mechanism of adsorption of NH₃ in MFM-300(Sc) was examined systematically using gas isotherms, breakthrough experiments, *in situ* solid-state nuclear magnetic resonance (ssNMR) spectroscopy, synchrotron infrared microspectroscopy, neutron powder diffraction (NPD) and inelastic neutron scattering (INS) techniques, coupled with DFT modelling. Distinct new insights have been gained into the mechanism of adsorption compared with a recent report based on theoretical and infrared spectroscopic studies of this system.¹⁷ Importantly, we found the exceptional NH₃ uptake (19.5 mmol g⁻¹ at 273 K and 1 bar) by MFM-300(Sc) was mediated by reversible host–guest and guest–guest hydrogen bond interactions. The moderate strength of the host–guest interaction in MFM-300(Sc) leads to excellent adsorption reversibility and stability with full retention of the capacity over 90 cycles.

MFM-300(Sc) shows a three-dimensional framework containing [ScO₄(OH)₂] octahedra which are connected *via* the

^a Department of Chemistry, University of Manchester, Manchester, M13 9PL, UK.
 E-mail: M.Schroder@manchester.ac.uk, Sihai.Yang@manchester.ac.uk

^b Department of Chemical Engineering and Analytical Science, University of Manchester, Manchester, M13 9PL, UK

^c ISIS Facility, STFC Rutherford Appleton Laboratory, Oxfordshire, OX11 0QX, UK

^d Neutron Scattering Division, Neutron Sciences Directorate, Oak Ridge National Laboratory, Oak Ridge, TN 37831, USA

^e Diamond Light Source, Harwell Science and Innovation Campus, Oxfordshire, OX11 0DE, UK

† Electronic supplementary information (ESI) available. CCDC 2142629, 2142630 and 2142631. For ESI and crystallographic data in CIF or other electronic format see DOI: <https://doi.org/10.1039/d2cc01197b>





Fig. 1 (a) Adsorption isotherms for NH_3 in MFM-300(Sc) at 273–313 K (adsorption: solid symbols; desorption: open symbols). (b) 90 cycles of adsorption–desorption of NH_3 in MFM-300(Sc) under pressure-swing conditions. (c) Breakthrough curve for NH_3 (1000 ppm diluted in He) through a fixed-bed packed with MFM-300(Sc) at 298 K and 1.0 bar. (d) Comparison of NH_3 uptake at 1 bar and 298 K for selected materials plotted against their surface areas (solid symbols: reversible sorption; hollow symbols: irreversible sorption; full details are given in the supplementary information).

cis- μ_2 -OH groups into infinite chains, and further coordinated by the BPTC⁴⁻ ligand (H_4BPTC = biphenyl-3,3',5,5'-tetracarboxylic acid) (Fig. S1, ESI[†]).¹⁸ Desolvated MFM-300(Sc) shows a Brunauer–Emmett–Teller (BET) surface area of $1390 \text{ m}^2 \text{ g}^{-1}$ and a pore volume of $0.48 \text{ cm}^3 \text{ g}^{-1}$ (Fig. S2, ESI[†]). MFM-300(Sc) exhibits high thermal stability up to $500 \text{ }^\circ\text{C}$ under N_2 (Fig. S3, ESI[†]) and high chemical stability in aqueous solutions of pH of 7–12 as well as in various organic solvents (Fig. S4, ESI[†]).

Adsorption isotherms of NH_3 for MFM-300(Sc) were measured at 273–313 K, where an exceptional uptake of 19.5 mmol g^{-1} was recorded at 273 K and 1.0 bar (Fig. 1a), reducing to 13.5 mmol g^{-1} at 298 K. MFM-300(Sc) shows the highest NH_3 uptake among the MFM-300(M) family^{6,14} primarily due to its large pore size and pore volume allowing the accommodation of additional NH_3 molecules in the pore. MFM-300(Sc) shows an NH_3 uptake of 13.5 mmol g^{-1} at 298 K and 1.0 bar, comparing favourably with state-of-the-art materials (Fig. 1d and Table S6, ESI[†]). The uptake of NH_3 in MFM-300(Sc) between 273 and 313 K decreases gradually with increasing temperature, consistent with an exothermic adsorption mechanism.¹⁹ The isosteric heat of adsorption (Q_{st}) for NH_3 in MFM-300(Sc) decreases from 60 to 30 kJ mol^{-1} with increasing loading of NH_3 from 1 to 10 mmol g^{-1} (Fig. S7, ESI[†]), confirming the presence of moderate adsorbate–adsorbent binding interaction. The repeated isotherm of NH_3 at 273 K using regenerated MFM-300(Sc) shows no loss in capacity with full retention of its porosity (Fig. S6, ESI[†]). 90 consecutive cycles of adsorption–desorption were conducted at 298 K and confirmed excellent reversibility and stability of adsorption (Fig. 1b), with retention of the crystal structure of MFM-300(Sc) as confirmed by powder X-ray diffraction (PXRD) (Fig. S5, ESI[†]). The ability of MFM-300(Sc) to capture NH_3 at low concentrations (1000 ppm)

was confirmed by dynamic breakthrough experiments at 298 K with a dynamic uptake of 1.65 mmol g^{-1} , consistent with that measured by isotherms (1.74 mmol g^{-1} at 10 mbar, equivalent to 1000 ppm; Fig. 1c). With an exceptional adsorption capacity and excellent regenerability, MFM-300(Sc) represents a promising candidate for applications in NH_3 storage and transport.

In situ NPD data of MFM-300(Sc) as a function of ND_3 loading were collected and Rietveld refinements revealed the preferential binding sites for ND_3 (Fig. 2). Interestingly, the NH_3 -induced rearrangement of metal–ligand (Sc–O) bonds *via* insertion of NH_3 molecules into the MOF upon ND_3 binding as predicted by a DFT study¹⁷ was not observed here. For MFM-300(Sc)·(ND_3)_{1.25}, $\{[\text{Sc}_2(\text{L})(\text{OD}_{0.6}\text{H}_{0.4})_2](\text{ND}_{2.05}\text{H}_{0.95})_{1.25}\}$, only one binding site was found, interacting primarily with the bridging μ_2 -OH groups at the four corners of its square-shaped channel [$\text{O}_{\text{bridge}}\text{-H}\cdots\text{ND}_3 = 1.96(1) \text{ \AA}$] (Fig. 2a and b). At the higher loading of MFM-300(Sc)·(ND_3)_{2.6}, $\{[\text{Sc}_2(\text{L})(\text{OD}_{0.75}\text{H}_{0.25})_2](\text{ND}_{2.42}\text{H}_{0.58})_{2.6}\}$, two distinct binding sites were identified (Fig. 2c and d). Site I is fully occupied by ND_3 molecules ($1 \text{ ND}_3/\text{Sc}$), with hydrogen bonding between the μ_2 -OH groups and the ND_3 molecule [$\text{O}_{\text{bridge}}\text{-H}\cdots\text{ND}_3 = 1.93(1) \text{ \AA}$]. This is complemented by additional electrostatic interactions [$\text{ND}_3\cdots\text{aromatic rings} = 3.13(1) \text{ \AA}$], and hydrogen bonding [$\text{ND}_3\cdots\text{O}_{\text{ligand}} = 3.24(1) \text{ \AA}$]. Site II ($0.3 \text{ ND}_3/\text{Sc}$) exhibited hydrogen bonding with the ND_3 at site I [$2.30(3) \text{ \AA}$ and $2.24(2) \text{ \AA}$], propagating along the 1D channel to form a cooperative $\{\text{ND}_3\}_\infty$ network. Similar to other MFM-300 analogues,^{6,14} hydrogen/deuterium site exchange was also observed between the adsorbed ND_3 molecules and the μ_2 -OH group for MFM-300(Sc).

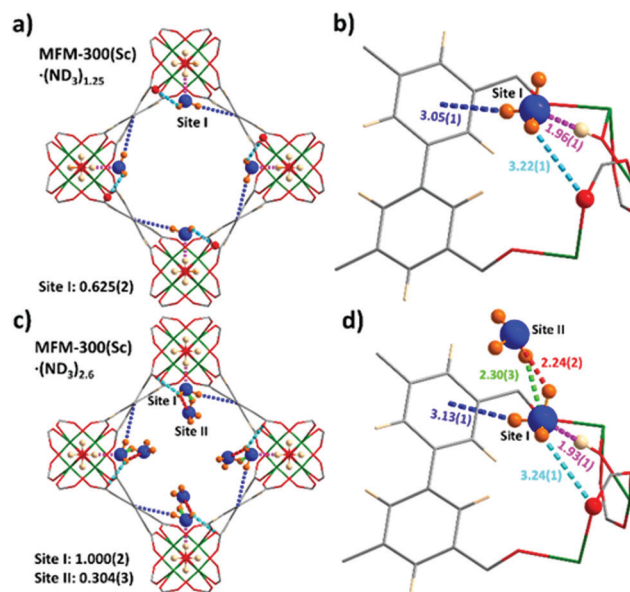


Fig. 2 Views of binding sites for ND_3 in MFM-300(Sc) determined by NPD at 10 K. The occupancy of each site has been converted into ND_3/Sc for clarity. (a and c) Views along the *c*-axis showing packing of the guest molecules of ND_3 in MFM-300(Sc)·(ND_3)_{1.25} and MFM-300(Sc)·(ND_3)_{2.6}, respectively. (b and d) Detailed views of host–guest interactions between MFM-300(Sc) and adsorbed molecules of ND_3 .



The analysis of the NPD data were entirely consistent with information from solid-state NMR spectroscopy. Upon loading MFM-300(Sc) with NH_3 , only slight structural modifications were observed, and the crystalline nature of the framework was retained. ^{45}Sc magic angle spinning (MAS) NMR spectroscopy confirmed that the geometry around the Sc(III) centre was not notably distorted by interaction with NH_3 , with the $\mu_2\text{-OH}$ groups (Fig. S11a, ESI †) and $\{^1\text{H}\}\text{-}^{13}\text{C}$ CPMAS NMR spectra showing that the carboxyl resonance from the linker is unaffected upon NH_3 loading (*i.e.* minimal metal site distortion). However, the resonances assigned to the aromatic carbons do shift, reflecting an interaction of the rings with the guest molecules (Fig. S11b, ESI †). The interaction of NH_3 with the MOF was also investigated using 2D $^1\text{H}\text{-}^{45}\text{Sc}$ dipolar correlation (HETCOR) NMR spectroscopy (Fig. 3). The spectrum of pristine MFM-300(Sc) (Fig. 3a) shows clear cross peaks between aromatic protons (from the linker), as well as from hydroxyl protons ($\mu_2\text{-OH}$ groups), with the Sc(III) site, demonstrating a close proximity between these atomic environments. The corresponding spectrum of NH_3 -loaded MFM-300(Sc) is substantially different. Whilst cross peaks with aromatic protons unchanged, cross peaks with $\mu_2\text{-OH}$ groups have moved to higher chemical shifts, indicating the presence of hydrogen-bonding, and a new weak cross peak is observed and assigned to pore-confined NH_3 protons. *In situ* synchrotron FTIR microspectra were recorded at 298 K (Fig. 3e and f). The characteristic

O–H stretching mode of the $\mu_2\text{-OH}$ group is observed at 3678 cm^{-1} , which reduces in intensity and broadens upon loading of NH_3 . The band at 3404 cm^{-1} is assigned to the N–H stretching of NH_3 , and this exhibits a red shift to 3390 cm^{-1} .^{6,14} The bands at 1614 and 1440 cm^{-1} , assigned to $\nu_{\text{as}}(\text{COO}^-)$ and $\nu_{\text{s}}(\text{COO}^-)$, respectively,²⁰ show small red shifts upon adsorption of NH_3 ($\Delta = 4 - 7\text{ cm}^{-1}$), consistent with interaction between NH_3 and carboxylate groups. The bands between 3800 and 1400 cm^{-1} for the local framework remain unchanged upon re-activation, confirming the high structural stability of MFM-300(Sc). Thus, the ssNMR and FTIR studies verify that NH_3 is hydrogen-bonded to the $\mu_2\text{-OH}$ groups *via* the lone pair of electrons on nitrogen, fully consistent with the NPD analysis.

INS spectra of bare and NH_3 -loaded MFM-300(Sc) were also collected (Fig. S12, ESI †) and simulated using DFT calculations based upon the structural models derived from NPD analyses (Fig. S13, ESI †). The difference spectra (Fig. 4a), which were obtained by subtracting the spectrum of the bare MOF from that of the NH_3 -loaded MOF, show the vibrational features of both the adsorbed NH_3 molecules and the changes for the MOF host. The peaks in the low energy region (Fig. 4b) are primarily due to the vibrational modes of adsorbed NH_3 molecules, with a small contribution due to changes in the lattice modes of the framework. The agreement between experimental and simulated spectra in terms of the overall profile allows unambiguous assignment of all major peaks. Specifically, the bands between 45 and 116 cm^{-1} are assigned to the translational motion of the NH_3 , which includes the vibration of NH_3 molecules perpendicular to and along the molecular C_3 axis and the hybrid of these modes. Peaks at 132 and 172 cm^{-1} are due to rotational motion of the NH_3 around its C_3 axis. Bands between 207 and 334 cm^{-1} are assigned to the rocking modes of the NH_3 . Compared to the spectrum of NH_3 in the solid state, where each NH_3 forms a 3D hydrogen bonding network with 6 adjacent NH_3 molecules, bands in all regions for the adsorbed NH_3 shift to lower energy and exhibit more broad features, indicating more dynamic environment for the adsorbed NH_3 . The features in the higher energy region mainly reflect the modes of the framework (Fig. 4c). Features I and III are due to the broadening of the peaks at 692 and 934 cm^{-1} for bare MFM-300(Sc), and are assigned to the C–H rocking out of the C_6 plane, in-phase and anti-phase, respectively. Feature VI at higher frequency between 1090 and 1163 cm^{-1} shows reduced intensity upon adsorption of NH_3 , and is assigned to the H_{ring} rocking within the C_6 plane. Features II and IV originate from a significant blue shift of the peak at 754 cm^{-1} in the spectrum of bare MFM-300(Sc) to 987 cm^{-1} in the spectrum of NH_3 -loaded MFM-300(Sc). This is assigned to the rocking of $\mu_2\text{-OH}$ within the Sc–O–Sc plane. Interestingly, the features involving the motions of H_{ring} show only broadening and a decrease in intensities, while the features involving the motion of the $\mu_2\text{-OH}$ experience changes in energy. This indicates a stronger interaction between NH_3 and the $\mu_2\text{-OH}$ than with H_{ring} centres. Feature V in the difference spectrum is due to the umbrella motion of adsorbed NH_3 . Thus, the combined INS and DFT study has visualised directly the host–guest binding dynamics,



Fig. 3 $^1\text{H}\text{-}^{45}\text{Sc}$ Heteronuclear dipolar correlation spectroscopy (HETCOR) MAS NMR spectra of (a, c) pristine and (b, d) NH_3 -loaded MFM-300(Sc), with corresponding ^{45}Sc MAS NMR spectra (top). The spectra were recorded at 9.4 T using a MAS frequency of 12 kHz. The dashed blue lines highlight correlations between the Sc(III) site and various proton environments. *In situ* FTIR spectra of MFM-300(Sc) as a function of NH_3 loading (diluted in dry N_2) and re-activated under a flow of dry N_2 at 100 mL min^{-1} at 298 K for 2 h: (e) $3800\text{--}3200\text{ cm}^{-1}$, (f) $1650\text{--}1400\text{ cm}^{-1}$.



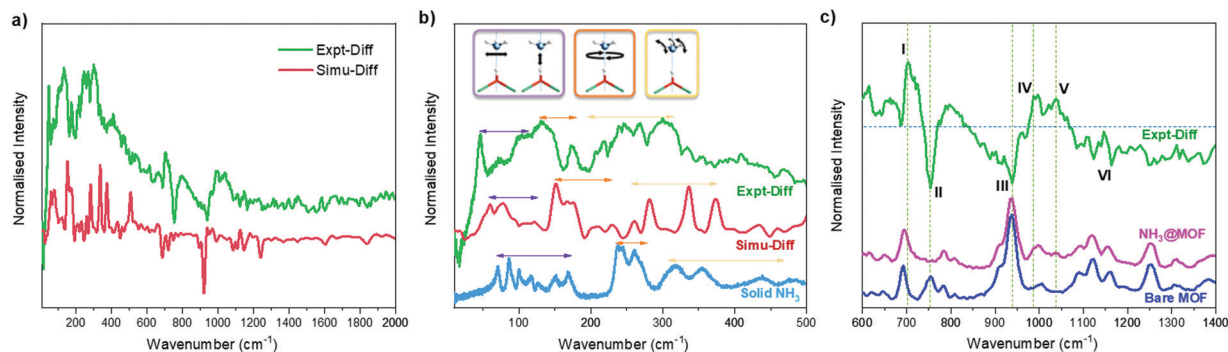


Fig. 4 Experimental and simulated INS difference spectra of the adsorbed NH_3 within MFM-300(Sc), denoted as Expt-Diff and Simu-Diff, respectively. (b) Comparison of the INS spectra of adsorbed NH_3 with solid NH_3 . (c) Experimental INS spectra of bare MFM-300(Sc), NH_3 -loaded MFM-300(Sc) and the difference spectrum at the higher energy region.

consistent with the reversible and high adsorption of NH_3 in MFM-300(Sc).

In summary, MFM-300(Sc) comprised of metal-oxide chains with bridging $-\text{OH}$ groups shows exceptional adsorption capacity (19.5 mmol g^{-1} at 273 K and 1 bar) and regenerability for NH_3 . *In situ* NPD analysis, ^{45}Sc ssNMR spectroscopy, synchrotron FTIR and INS/DFT studies have unambiguously visualised the binding interactions and dynamics of NH_3 within the pores of MFM-300(Sc). This in-depth understanding of the structure–function relationship with these host–guest systems will enable the rational design of potential materials with desired properties.

We thank EPSRC (grant EP/I011870), The Royal Society, and University of Manchester for funding. This project has received funding from the European Research Council (ERC) under the European Union's Horizon 2020 research and innovation programme (grant agreement No 742401, NANO-CHEM). We are grateful to STFC/ISIS facility for access to Beamlines TOSCA and WISH. We thank Diamond Light Source for access to Beamline B22. LG and YM thank the China Scholarship Council (CSC) for funding. The computing resources were made available through the VirtuES and the ICE-MAN projects, funded by Laboratory Directed Research and Development program and Compute and Data Environment for Science (CADES) at ORNL.

Conflicts of interest

The authors declare no competing interest.

Notes and references

- 1 D. R. MacFarlane, P. V. Cherepanov, J. Choi, B. H. R. Suryanto, R. Y. Hodgetts, J. M. Bakker, F. M. Ferrero Vallana and A. N. Simonov, *Joule*, 2020, **4**, 1186–1205.
- 2 S. P. Mirajkar, B. S. Rao, M. J. Eapen and V. P. Shiralkar, *J. Phys. Chem. B*, 2001, **105**, 4356–4367.
- 3 T. Zeng, H. Huang, N. Kobayashi and J. Li, *Nat. Resour.*, 2017, **08**, 611–631.

- 4 C. J. Doonan, D. J. Tranchemontagne, T. G. Glover, J. R. Hunt and O. M. Yaghi, *Nat. Chem.*, 2010, **2**, 235–238.
- 5 H. Demir, K. S. Walton and D. S. Sholl, *J. Phys. Chem. C*, 2017, **121**, 20396–20406.
- 6 X. Han, W. Lu, Y. Chen, I. da Silva, J. Li, L. Lin, W. Li, A. M. Sheveleva, H. G. W. Godfrey, Z. Lu, F. Tuna, E. J. L. McInnes, Y. Cheng, L. L. Daemen, L. J. M. McPherson, S. J. Teat, M. D. Frogley, S. Rudic, P. Manuel, A. J. Ramirez-Cuesta, S. Yang and M. Schröder, *J. Am. Chem. Soc.*, 2021, **143**, 3153–3161.
- 7 T. N. Nguyen, I. M. Harreschou, J. H. Lee, K. C. Stylianou and D. W. Stephan, *Chem. Commun.*, 2020, **56**, 9600–9603.
- 8 C. Marsh, X. Han, J. Li, Z. Lu, S. P. Argent, I. da Silva, Y. Cheng, L. L. Daemen, A. J. Ramirez-Cuesta, S. P. Thompson, A. J. Blake, S. Yang and M. Schröder, *J. Am. Chem. Soc.*, 2021, **143**, 6586–6592.
- 9 S. Yang, J. Sun, A. J. Ramirez-Cuesta, S. K. Callear, W. I. David, D. P. Anderson, R. Newby, A. J. Blake, J. E. Parker, C. C. Tang and M. Schröder, *Nat. Chem.*, 2012, **4**, 887–894.
- 10 G. Smith, J. Eyley, X. Han, X. Zhang, J. Li, N. Jacques, H. Godfrey, S. Argent, L. McPherson, S. Teat, Y. Cheng, M. Frogley, G. Cinque, S. Day, C. Tang, T. Easun, S. Rudic, A. Cuesta, S. Yang and M. Schröder, *Nat. Mater.*, 2019, **18**, 1358–1365.
- 11 D. Saha and S. Deng, *J. Colloid Interface Sci.*, 2010, **348**, 615–620.
- 12 A. J. Rieth and M. Dincă, *J. Am. Chem. Soc.*, 2018, **140**, 3461–3466.
- 13 A. J. Rieth, Y. Tulchinsky and M. Dincă, *J. Am. Chem. Soc.*, 2016, **138**, 9401–9404.
- 14 H. G. W. Godfrey, I. da Silva, L. Briggs, J. H. Carter, C. G. Morris, M. Savage, T. L. Easun, P. Manuel, C. A. Murray, C. C. Tang, M. D. Frogley, G. Cinque, S. Yang and M. Schröder, *Angew. Chem., Int. Ed.*, 2018, **57**, 14778–14781.
- 15 Y. Chen, F. F. Zhang, Y. Wang, C. Y. Yang, J. F. Yang and J. P. Li, *Microporous Mesoporous Mater.*, 2018, **258**, 170–177.
- 16 D. W. Kim, D. W. Kang, M. Kang, J. H. Lee, J. H. Choe, Y. S. Chae, D. S. Choi, H. Yun and C. S. Hong, *Angew. Chem., Int. Ed.*, 2020, **132**, 22720–22725.
- 17 P. Lyu, A. M. Wright, A. L. Olvera, P. G. M. Mileo, J. A. Zárate, E. M. Ahumada, V. Martis, D. R. Williams, M. Dincă, I. A. Ibarra and G. Maurin, *Chem. Mater.*, 2021, **33**, 6186–6192.
- 18 X. Zhang, I. da Silva, H. G. W. Godfrey, S. K. Callear, S. A. Sapchenko, Y. Cheng, I. Vitorica-Yrezabal, M. D. Frogley, G. Cinque, C. C. Tang, C. Giacobbe, C. Dejoie, S. Rudic, A. J. Ramirez-Cuesta, M. A. Denecke, S. Yang and M. Schröder, *J. Am. Chem. Soc.*, 2017, **139**, 16289–16296.
- 19 S. Yang, L. Liu, J. Sun, K. M. Thomas, A. J. Davies, M. W. George, A. J. Blake, A. H. Hill, A. N. Fitch, C. C. Tang and M. Schröder, *J. Am. Chem. Soc.*, 2013, **135**, 4954–4957.
- 20 K. I. Hadjiivanov, D. A. Panayotov, M. Y. Mihaylov, E. Z. Ivanova, K. K. Chakarova, S. M. Andonova and N. L. Drenchev, *Chem. Rev.*, 2021, **121**, 1286–1424.

