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Synthesis of high-molecular-weight aliphatic polycarbonates by organo-catalysis†

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Aliphatic polycarbonates have attracted significant attention for biomedical application over the last few years due to their biodegradability, low toxicity and good biocompatibility. However, in most cases, the use of metal-based catalysts is required for the preparation of aliphatic polycarbonates by the polycondensation method, which is difficult to remove completely from the final polymer. For this reason, our work is focused on the synthesis of high-molecular-weight aliphatic polycarbonates using organo-catalysts *via* a two-step polycondensation of dimethyl carbonate and a linear alkane diol as monomers. A variety of organo-catalysts have been surveyed for the synthesis of aliphatic polycarbonates. The influence of thiourea with mono- or bi-electron acceptor groups as cocatalysts, which were found to activate the carbonyl groups of lactide and trimethylene carbonate in the ring opening polymerization successfully, was investigated in the polycondensation. In summary, high-molecular-weight aliphatic polycarbonates, such as poly(1,4-butylene carbonate) (PBC), poly(1,5-pentamethylene carbonate) (PPC) and poly(1,6-hexamethylene carbonate) (PHC), were successfully prepared with number averaged molar masses (M_n) up to 23 000 g mol⁻¹, dispersities below 1.8 and high yield of >80% under relatively mild operating conditions ($T < 130$ °C) using 4-dimethylaminopyridine (DMAP) as the catalyst. At 170 °C the poly(1,4-butylene carbonate) with M_n of 52 000 g mol⁻¹ was synthesized. Additionally, hydroxyl group terminated poly(1,4-butylene carbonate) with M_n up to 17 000 g mol⁻¹ was obtained and characterized by ¹H NMR spectroscopy and ESI-TOF-mass spectrometry. The ratio of end groups (–OH/–OC(O)O–CH₃) could be adjusted by using different feed ratios or catalysts.

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Introduction

Polycarbonates (PCs) are polymers containing repeating carbonate groups ($-\text{O}-(\text{C}=\text{O})-\text{O}-$). Aromatic polycarbonates are widely used as engineering plastics^{1,2} because of their attractive mechanical properties, *e.g.* low moisture absorption, high impact strength, high elastic modulus, creep resistance and good thermal stability. However, compared with traditional aromatic polycarbonates, aliphatic polycarbonates have received little attention because of their poor thermal stability and high susceptibility to hydrolysis.^{1,3–7} Over the last few years, aliphatic polycarbonates have attracted significantly increasing attention for biomedical application, *e.g.* for the composition of biomedical implants and acting as drug delivery devices, due to their biodegradability, low toxicity and good biocompatibility.^{1,4,8–11} Aliphatic polycarbonates can be prepared through different methods, such as copolymerization

of CO₂ and an epoxide (Scheme 1a),^{12,13} which is only suitable for the synthesis of aliphatic polycarbonates in which the carbonate linkages are connected by two carbon atoms, ring opening polymerization of cyclic carbonate monomers (Scheme 1b)^{1,14–18} and condensation polymerization of dialkyl- or diphenyl carbonate and aliphatic diols (Scheme 1c).^{4,19–26}

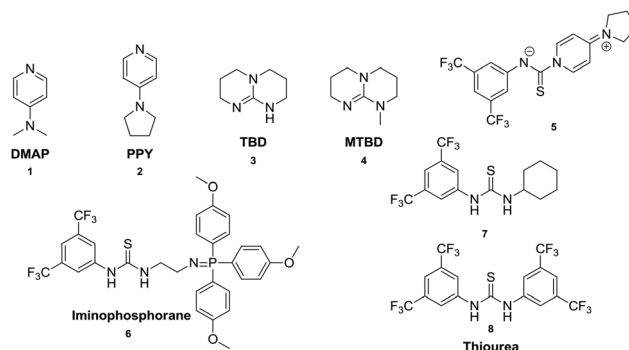
The ring opening polymerization of cyclic carbonates is one of the most effective methods to obtain polycarbonates with a high molar mass and low dispersity.²⁰ However, cyclic carbonate monomers are very expensive because of their low synthetic yields. Hence, polycarbonates from ring opening polymerization have been mainly investigated for biomedical application.^{22,27,28} The best strategy for large-scale preparation of aliphatic polycarbonates is the two-step condensation polymerization of dimethyl carbonate (DMC) and aliphatic diols with more than three carbon atoms. Oligomers with a molar mass lower than 1000 g mol⁻¹ are obtained in the first, initial condensation step, due to the low equilibrium constant. In the second step, polymer chains propagated by transesterification between the hydroxyl and methyl carbonate or two methyl carbonate end groups in the presence of transesterification catalysts, while high temperature and high vacuum are required to remove unreacted monomers and freshly generated byproducts.

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The resulting oligomers or polymers have three possible end group compositions, hydroxyl end groups, methyl carbonate end groups or a combination of both (Scheme 1c).^{19,20,22} Recently, Li *et al.* have reported the preparation of polycarbonates with a high molar mass (M_n up to 94 000 g mol⁻¹) using a novel TiO₂/SiO₂-poly(vinyl pyrrolidone)-based catalyst (TSP-44).²¹ Lee *et al.* have used NaH as the catalyst to prepare aliphatic polycarbonates with a high molar mass (M_n up to 150 000 g mol⁻¹) successfully with the prerequisite that the [–OH]/[–OCH₃] ratio of the oligomers generated in the transesterification step is about 1.0.²² However, in most cases, the use of metal-based catalysts is required for the preparation of aliphatic polycarbonates by the polycondensation method, which is difficult to remove completely from the final polymer.

For this reason, our work is focused on the synthesis of high-molecular-weight polycarbonates using organo-catalysts *via* a two-step polycondensation of dimethyl carbonate and a linear alkane diol as monomers. Some organo-catalysts such as guanidines, amidines and tertiary amines have been used in the ring opening polymerization of trimethylene carbonate (TMC) and shown to yield poly(trimethylene carbonate) with a high molar mass (M_n up to 72 000 g mol⁻¹), with low dispersities ($D_M = 1.04\text{--}1.80$) and with well-defined terminal groups.^{15,16,29} Furthermore, thiourea derivatives have been reported for the direct activation of electrophilic substrates *via* employment of double hydrogen bonding. Hedrick^{30,31} and Dixon³² demonstrated that thiourea based bifunctional organo-catalysts effectively activated the ring opening polymerization of cyclic esters. Moreover, Hedrick reported that electrophilic thioureas and nucleophilic bases are not required to be linked in the same molecule.³¹ Kosugi has exploited a 3,5-bis(trifluoromethyl)phenyl and 4-pyrrolidinopyridine (PPY) based zwitter ionic salt organo-catalyst for transesterification



Scheme 2 Organo-catalysts for the synthesis of aliphatic polycarbonates *via* the polycondensation method.

reactions.³³ However, there have been few reports about the successful synthesis of aliphatic polycarbonates with high-molecular-weight using organo-catalysts through condensation polymerization of DMC and diols. Picquet and Plasseraud described a route to the synthesis of aliphatic polycarbonates (M_n up to 7400 g mol⁻¹) using 1-*n*-butyl-3-methylimidazol-2-carboxylate (BMIM-2-CO₂) as a catalyst.⁴

In this work, a variety of organo-catalysts (Scheme 2) have been surveyed for the synthesis of aliphatic polycarbonates. The influence of thiourea with mono- or bi-electron acceptor groups as cocatalysts, which were found to activate the carbonyl groups of lactide and trimethylene carbonate in the ring opening polymerization successfully, was investigated in the polycondensation as well.

Experimental

Materials

1,4-Butane diol (99+%) was purchased from Acros Organics and vacuum distilled using a short path distillation apparatus and dried with 4 Å molecular sieves before use. Other diols ($C > 4$) were dried under vacuum at ambient temperature overnight before use. The catalysts arylaminothiocarbonylpyridinium salt (5),³³ bifunctional iminophosphorane (6)³² and thioureas (7 and 8)³⁴ were synthesized as previously reported. Other reagents were commercially available and used as received.

Measurements

¹H and ¹³C NMR spectra were recorded by using a Bruker AV 500 spectrometer at 500 MHz and 125 MHz, respectively. Chloroform-*d* (CDCl₃, 99.8 D%), dimethylsulfoxide-*d*₆ (DMSO-*d*₆, 99.5 D%) or acetonitrile-*d*₃ (MeCN-*d*₃, 99 D%) were used as solvents for NMR measurements. The molar masses and molar mass distribution (M_w/M_n) were analyzed by employing a size exclusion chromatography (SEC) system equipped with four consecutive columns (PSS-SDV columns filled with 5 μm gel particles with a defined porosity of 10⁶ Å, 10⁴ Å, 10³ Å and 10² Å, respectively) and a Shodex RI-detector (RI-101) at 30 °C.

	[Diol]:[DMC]:[cat.]	Cat.	T (2 step) (°C)	Yields (%)	M_n^a (g mol ⁻¹)	D_M^a	End groups ^b [-OCH ₃]:[-OH]
PBC 1	1:1.5:0.005	Cat. 5	130	60	5900	1.85	2:98
PBC 2	1:1.5:0.01	Cat. 5	130	70	9000	1.69	14:86
PBC 3	1:2.0:0.01	Cat. 5	130	65	11 000	1.71	80:20
PBC 4	1:1.2:0.005	Cat. 1	130	57	16 000	1.66	0:100
PBC 5	1:1.2:0.005	Cat. 2	130	59	7900	2.03	0:100
PBC 6	1:1.2:0.005	Cat. 4	130	61	17 000	1.77	0:100
PBC 7	1:1.2:0.005	Cat. 5	130	57	4100	2.40	0:100
PBC 8	1:1.2:0.005	Cat. 6	130	53	13 000	1.68	0:100
PBC 9	1:2.0:0.01	Cat. 1	130	85	23 000	1.77	32:68
PBC 10	1:2.0:0.01	Cat. 1	170	79	52 000	1.77	70:30
PPC 1	1:2.0:0.01	Cat. 1	130	77	22 000	1.60	43:57
PHC 1	1:2.0:0.01	Cat. 1	130	88	23 000	1.53	61:39

To determine the influence of polymerization times on the molar mass of PBC, a kinetic study of PBC 9 ($[\text{BD}]:[\text{DMC}]:$

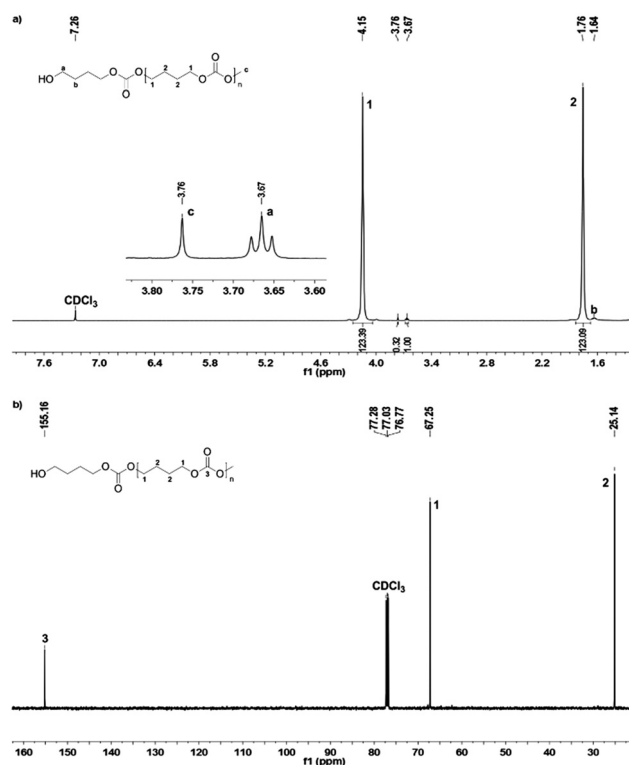


Fig. 2 ^1H (a) and ^{13}C NMR (b) spectra of PBC 9.

[DMPA] = 1 : 2 : 0.01, 130 °C) was carried out. Fig. 3 shows the molar mass and molar mass distribution data determined by SEC. The molar mass of the polymer increased rapidly throughout the initial 30 min. After a reaction time of 3 h, a molar mass of 14 000 g mol⁻¹ was obtained. When the condensation reaction was further conducted, the molar mass increased slowly and finally up to $M_n = 23\,000$ g mol⁻¹ for 24 h reaction time. The dispersity values remained below 1.8 during the condensation reaction.

Hydroxyl terminated PBCs (PBC 4–9 in Table 2) have also been investigated by ESI-TOF-MS to determine the end groups.

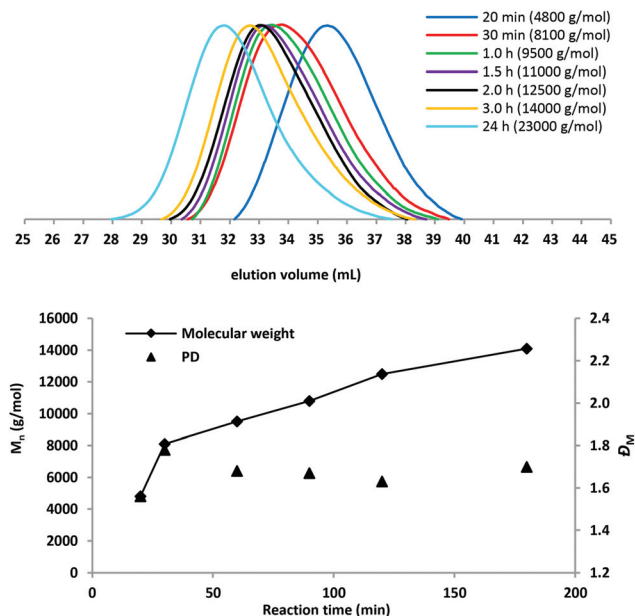


Fig. 3 SEC traces and plot M_n (determined by SEC) and the dispersity values of PBC 9 versus the polymerization time in the 2nd step.

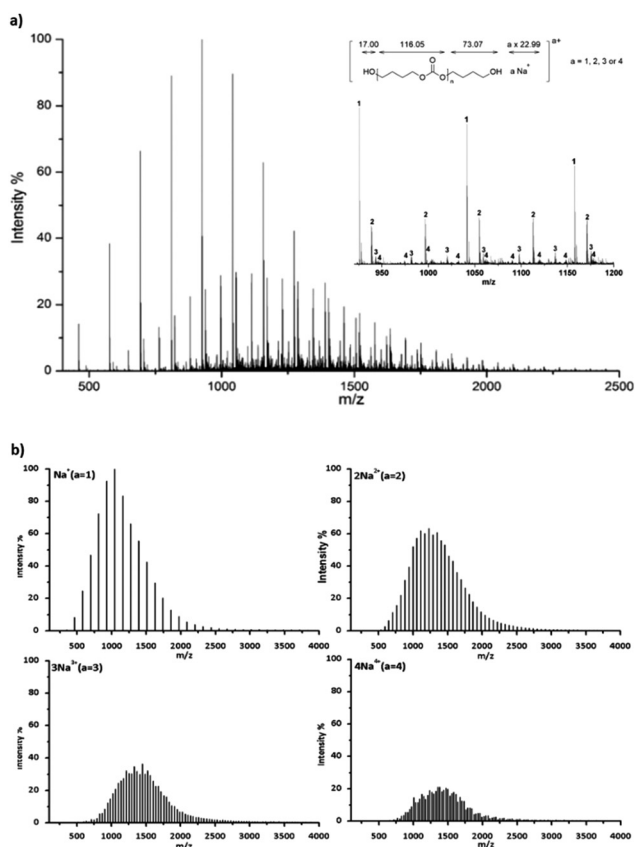


Fig. 4 ESI-TOF mass spectrum of PBC 7 (Table 2, entry 7) terminated with the hydroxyl group, (a) complete spectrum and a part of the spectrum distinguished by carrying charges $a = 1, 2, 3$ and 4 in the region m/z 400 to 2500, and (b) separated spectra with $a = 1, 2, 3$ and 4 .

Table 3 Calculated and experimental m/z in the ESI-TOF mass spectrum of PBC 7 with different charges (a up to 4)

a	Calculated [Da]	Found [Da]	Calculated [Da]	Found [Da]	Difference [Da]
1	1042.01	1042.19 ($n = 8$)	1158.12	1158.33 ($n = 9$)	116.14
2	1229.19	1229.36 ($n = 20$)	1287.24	1287.37 ($n = 21$)	58.01
3	1214.18	1214.38 ($n = 30$)	1252.88	1253.06 ($n = 31$)	38.68
4	1206.67	1206.86 ($n = 40$)	1235.69	1235.91 ($n = 41$)	29.05

Fig. 4a shows a typical ESI-TOF-MS spectrum for hydroxyl terminated PBC. Polymers were multiply charged during the ionization. The different series can be separated by Ion Mobility Separation (IMS).³⁵ The separated spectra of up to tetra charged polymers are shown in Fig. 4b. Moreover, the penta and hexa charged polymers were also detected but they are distributed with low intensity. The ESI-TOF-MS spectra show the presence of the main series of polymer chains corresponding to (HO-PBC- C_4H_8OH $a \times Na^+$) ($a = 1, 2, 3$ or 4) with repeating units of 116.05 Da, which is the molar mass of the repeating PBC unit. For the doubly charged polymer with $n = 20$ (Table 3, entry 2), the measured value of 1229.36 Da corresponded to the calculated value of 1229.19 Da using eqn (2). No further series could be seen, indicating that the polymer was only terminated with hydroxyl groups at both the chain ends. Hence, the organo-catalyzed synthesis of polycarbonates proceeded successfully without any side reaction, such as decarboxylation.

$$m/z = \frac{M(BD) + M(\text{monomer unit}) \times n + a \times Na^+}{a} \quad (2)$$

The thermal properties of the PBC, PPC and PHC samples were evaluated by DSC as shown in Table 4. The PBC samples displayed glass transition temperatures (T_g) of -36 – 31 °C and T_g increases with the increasing molar mass. The T_g of the PHC sample tended to lower T_g due to the higher chain flexibility. The melting temperatures (T_m) were observed at 56 – 62 °C, while the PPC and PHC showed lower T_m . In our case, the T_g was not visibly affected by the nature of the chain end group compositions.

Table 4 Thermal properties of aliphatic polycarbonate samples

	M_n^a (g mol ⁻¹)	T_g^b (°C)	T_m^b (°C)
PBC 5	7900	^c	59.9
PBC 4	16 000	−35.9	61.6
PBC 9	23 000	−33.2	59.0
PBC 10	52 000	−31.9	56.4
PPC 1	22 000	−42.4	54.2
PHC 1	23 000	−38.6	55.5

^a Determined using SEC in chloroform with PS standards. ^b T_g and T_m were measured by DSC. ^c Not detected.

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