



Cite this: *Sustainable Energy Fuels*,
2025, 9, 5673

Enhancing pH-gradient microscale bipolar interfaces (PMBI) enabled direct methanol hydrogen peroxide fuel cell (DMHPFC) performance under varying operating conditions

Kritika Sharma,^a Shrihari Sankarasubramanian,^{bc} Zhongyang Wang^d
and Vijay Ramani ^{*a}

This study introduces a direct methanol hydrogen peroxide fuel cell (DMHPFC) using a pH-gradient-enabled microscale bipolar interface (PMBI) to address limitations in direct methanol fuel cells (DMFCs). Unlike conventional fuel cells that use oxygen, the DMHPFC utilizes H₂O₂, enhancing reactant availability and reaction kinetics. The PMBI maintains separate pH environments at the anode and cathode. The PMBI-DMHPFC combines an alkaline anode for methanol oxidation and an acidic cathode for hydrogen peroxide reduction, achieving a theoretical open-circuit voltage (OCV) of 1.72 V (compared to a theoretical OCV of 1.25 V for DMFCs) and a volumetric energy density of 9.2 kWh l⁻¹ using aqueous methanol (39% vol) and hydrogen peroxide (41% vol). This energy density quadruples that of compressed hydrogen (2.1 kWh l⁻¹ at 69 MPa). This study identifies optimal operating conditions: 5 M methanol with 3 M KOH as anolyte, 5 M hydrogen peroxide with 1.5 M sulfuric acid as catholyte, Nafion 115 (127 μm) as membrane, and flow rate of 2.5 ml min⁻¹ cm⁻² – that maximize the power output and minimize activation-, ohmic- and mass transfer losses in DMHPFCs. Performance evaluation reveals a measured OCV of 1.69 V. While the PMBI-DMHPFC surpasses DMFC performance, its high OCV and energy density are not fully translated into high power density due to significantly higher activation and mass transport losses compared to H₂-O₂ fuel cells, which typically achieve peak power densities above 1000 mW cm⁻². The DMHPFC achieves a peak power density of 630 mW cm⁻² at the unusually high voltage of 0.8 V, reflecting the unique PMBI design and optimized operating conditions that reduce losses. This steeper voltage drop is attributed to sluggish reaction kinetics, membrane crossover and mass transport limitations. It highlights the potential for improved performance through advanced electrocatalysts, optimized membrane materials and flow design from this promising baseline.

Received 30th July 2025
Accepted 26th August 2025

DOI: 10.1039/d5se01042j

rs.c.li/sustainable-energy

1 Introduction

Selecting an appropriate fuel and oxidant is fundamental to ensuring fuel cells' technical and economic viability. Hydrogen-fed polymer electrolyte membrane fuel cells (H₂-PEMFCs), which use hydrogen as fuel and oxygen as the oxidant, are a well-established and widely studied technology.¹ However, their dependence on hydrogen is constrained by the lack of adequate infrastructure for its storage and distribution in both

liquid and gaseous forms.^{2,3} As an alternative, direct liquid fuel cells (DLFCs) have gained attention due to their ease of handling liquid fuels and compatibility with existing infrastructure.⁴ Among DLFCs,^{5–7} direct methanol fuel cells (DMFCs) are particularly notable. Methanol's high energy density (4.3 kWh l⁻¹), good electrochemical activity, widespread availability with annual global production exceeding 79 million metric tons (projected to approach 100 million metric tons by 2030),⁸ biodegradability, and low cost—currently around \$350–\$450 per metric ton,⁹ make it an attractive fuel option.¹⁰ It can also be directly fed into the anode, without requiring a reforming process, simplifying the system.¹¹ While toxic, its risks are manageable in commercial applications.¹² However, unlike H₂-PEMFCs, where kinetic losses are primarily at the cathode, DMFCs experience low power density due to sluggish kinetics at both electrodes (methanol oxidation reaction (MOR)¹³ and oxygen reduction reaction (ORR)¹⁴).

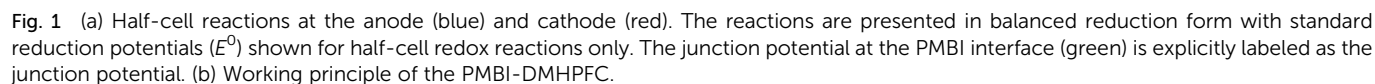
^aDepartment of Energy, Environmental and Chemical Engineering, Washington University in St. Louis, One Brookings Dr., St. Louis, MO 63130, USA. E-mail: ramani@wustl.edu

^bDepartment of Biomedical Engineering and Chemical Engineering, The University of Texas at San Antonio, San Antonio, TX 78249, USA

^cDepartment of Mechanical, Aerospace and Industrial Engineering, The University of Texas at San Antonio, San Antonio, TX 78249, USA

^dDepartment of Chemical and Biological Engineering, University of Alabama, Tuscaloosa, AL 35487, USA

alkaline media exhibit faster reaction rates than in acidic media, with activity decreasing in the order of $\text{NaOH} > \text{Na}_2\text{CO}_3 > \text{NaHCO}_3$.¹⁹ In contrast, acidic media is preferred for H_2O_2 due to reduced ionization-induced decomposition.²⁰ The theoretical potentials for the MOR and HPRR depend on the pH of the medium and are referenced *versus* the Standard Hydrogen Electrode (SHE) with explicit pH values. Specifically, the theoretical potential for MOR is -0.78 V vs. SHE at pH 14 (alkaline media) and -0.02 V vs. SHE at pH 0 (acidic media).²¹ In contrast, the theoretical potential for HPRR is 1.77 V vs. SHE at pH 0 (acidic media) and 0.87 V vs. SHE at pH 14 (alkaline media).²¹ Combining an alkaline anode for MOR with an acidic cathode for HPRR offers a promising configuration for achieving the highest possible theoretical open circuit voltage (OCV). As shown in Fig. 1a, this configuration yields an OCV of 1.72 V after accounting for the bipolar junction potential from water dissociation (-0.83 V).^{22,23} The bipolar junction potential arises from the interfacial ion transfer resistance between the acidic



and alkaline environments. The key to this configuration is maintaining distinct pH environments, which can be achieved here using a pH-gradient-enabled microscale bipolar interface (PMBI).²⁴ Previously explored in direct borohydride hydrogen peroxide fuel cells (DBHPFCs),^{24–26} the PMBI function is to maintain a localized alkaline environment at the anode even as bulk conditions near the anode remain non-alkaline due to the acidity of the cathode. This approach ensures sustained MOR activity and mitigates performance degradation caused by pH imbalance.

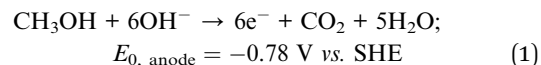
In addition to high theoretical OCV, a DMHPFC's energy density is higher than H₂-PEMFCs. Current state-of-the-art fuel cells typically store H₂ gas at 69 MPa. The volume-specific energy density for H₂ gas stored at 69 MPa is 2.1 kWh l⁻¹.²⁷ In contrast, aqueous methanol (39% by volume) and aqueous hydrogen peroxide (41% by volume) correspond to a volume-specific energy density of approximately 9.2 kWh l⁻¹,²⁷ comparable to gasoline, which also contains about 9.2 kWh l⁻¹ of available bond energy. This makes the DMHPFC's energy density per unit volume nearly four times greater than that of H₂-PEMFC, offering a compelling advantage for applications requiring compact energy storage. However, despite its high energy density, the DMHPFC is currently constrained by its low power density. While H₂-PEMFCs achieve peak power density (PPD) of approximately 1000 mW cm⁻²,²⁸ prior DMHPFC reaches only about 125 mW cm⁻² (ref. 29)—nearly 8 times lower. This disparity necessitates larger, heavier stacks to match power output, limiting their viability in weight- and size-sensitive applications like portable electronics or transportation systems.

This work develops a PMBI-DMHPFC, achieving an OCV of 1.69 V and PPD of approximately 630 mW cm⁻² at 0.8 V, exceeding the OCVs and PPDs of previous DMFC designs. Furthermore, the effects of operating conditions, including anolyte and catholyte concentrations, membrane thickness, and flow rate on the DMHPFC performance, are systematically explored to reveal the underlying performance characteristics of the PMBI-DMHPFC system. This exploration aims to identify optimal operating conditions to maximize efficiency and guide future performance-enhancing strategies.

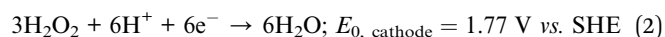
2 Working principle

The PMBI-DMHPFC, illustrated in Fig. 1b, consists of an alkaline anode compartment and an acidic cathode compartment separated by a PMBI. The anode comprises a catalyst layer (ACL), where the MOR occurs, a diffusion layer (ADL) for efficient methanol transport, and a flow field (AFF) to ensure uniform methanol distribution. Similarly, the cathode comprises a catalyst layer (CCL) for the HPRR, a diffusion layer (CDL), and a flow field (CFF). PMBI comprises the proton exchange membrane (PEM) intimately interfaced with a thin layer of anion exchange ionomer (AEI) binder at the ACL, which plays a critical role in maintaining localized alkaline conditions at the anode while preventing pH crossover.²⁴ During operation, an aqueous MeOH and hydroxide (OH⁻) solution is supplied to the anode, flowing through the AFF and diffusing through the

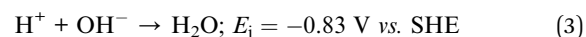
ADL to reach the ACL. At the ACL, methanol undergoes oxidation in the presence of OH⁻ according to the reaction:



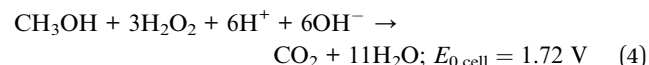
The generated electrons are then transported through an external circuit to the cathode, providing electrical power to the load. The resulting carbon dioxide (CO₂) and water enter the AFF before exiting the system. Simultaneously, a solution containing H₂O₂ and protons (H⁺) is supplied to the cathode, flowing through the CFF, and diffusing through the CDL to reach the CCL. At the CCL, H₂O₂ is reduced according to the reaction:



At the PMBI, water dissociation occurs, generating H⁺ ions, which diffuse toward the cathode and OH⁻ ions, which diffuse toward the anode under the influence of the electric field. This mechanism ensures the maintenance of distinct pH environments at each electrode.²² This process is driven by the electrochemical potential gradient across the membrane and occurs at the PMBI interface rather than the electrodes. This process prevents pH crossover while allowing efficient ion transport.



The overall cell reaction is:



As shown in Fig. 1, the theoretical OCV for this PMBI-DMHPFC is calculated as $E_{0, \text{cell}} = E_{0, \text{cathode}} - E_{0, \text{anode}} + E_j = 1.72 \text{ V}$.

The overall cell reaction (eqn (4)) demonstrates the theoretical potential for high energy conversion in the PMBI-DMHPFC. However, achieving this potential in practice requires overcoming substantial overpotential barriers at both electrodes. At anode, the sluggish kinetics of MOR can be partially mitigated by using high loadings of precious metal catalysts, particularly platinum (Pt), albeit at a significant cost increase.³⁰ During MOR, carbon monoxide (CO) forms as a reaction intermediate and strongly adsorbs on the Pt catalyst surface,³¹ blocking active sites and impeding further reaction. Mitigating CO poisoning requires a higher anodic overpotential to oxidize adsorbed CO into CO₂, thereby freeing active sites for MOR.³² To mitigate CO poisoning, binary alloys have been employed, with Pt–Ru outperforming others due to its superior CO tolerance and catalytic activity.³³ The Ru component provides oxygenated species (OH) at lower potentials,³⁴ which react with CO adsorbed on adjacent Pt sites, oxidizing it to CO₂. This bifunctional mechanism,³⁴ coupled with Ru's modification of Pt's electronic structure,^{35,36} weakens the Pt–CO bond³⁷ and enhances CO removal. While other binary alloys (e.g., Pt–Ni, Pt–Mo, Pt–Au, Pt–TiO₂) show



This journal is © The Royal Society of Chemistry 2025

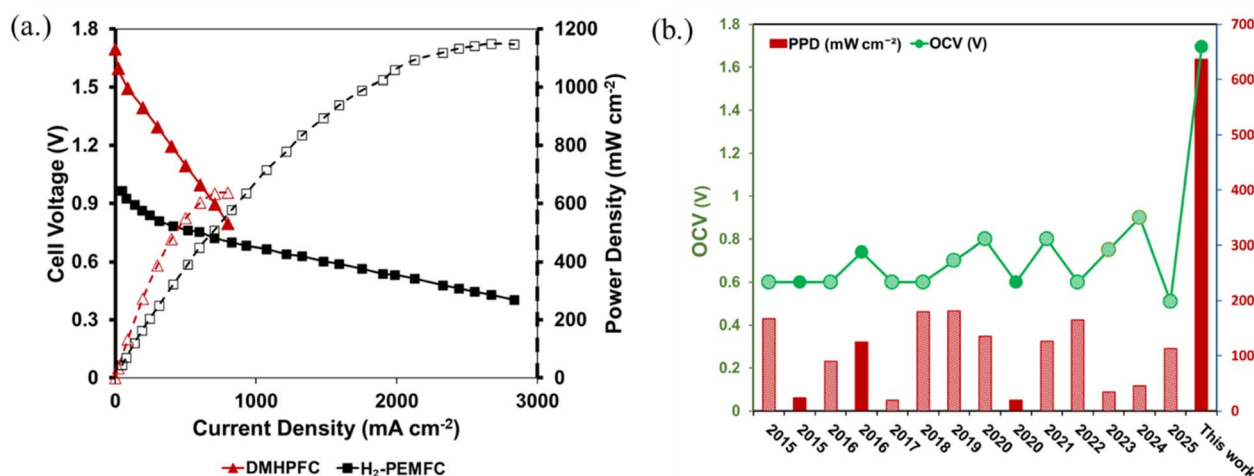


Fig. 2 (a) Polarization and power density curve of PMBI-DMHPFC and a high-power density H₂-PEMFCs. (b) Historical trend in DMFCs with oxidant as O₂ (shaded fill) and H₂O₂ (solid fill). The data is shown in Table S4.

thermodynamics of an alkaline anode medium and an acidic cathode medium. It should be noted that the observed high cell voltage is not attributable to a simple concentration cell mechanism. For a cell with only a pH gradient and hydrogen evolution/oxidation reactions occurring at the electrodes (a concentration cell), the maximum thermodynamic voltage is $\Delta E = 0.059 \text{ V} \times \Delta \text{pH}$, which is $\sim 0.83 \text{ V}$ for pH 0 to pH 14. In contrast, our system achieves an OCV of 1.69 V—well above this value—because the cell utilizes distinct redox couples: methanol oxidation at the alkaline anode ($E_{0,\text{anode}} = -0.78 \text{ V}$ vs. SHE at pH 14) and hydrogen peroxide reduction at the acidic cathode ($E_{0,\text{cathode}} = +1.77 \text{ V}$ vs. SHE at pH 0). These reactions are individually favorable and, when coupled, enable a high cell voltage with concomitant energy conversion, demonstrating that the cell is fundamentally a fuel cell, not a concentration cell. However, it is noteworthy that the measured OCV falls short of the theoretical value, which can be attributed to H₂O₂ decomposition, and MeOH and H₂O₂ crossover. The PMBI-DMHPFC's polarization curve demonstrated a steeper voltage drop than H₂-PEMFC, indicating greater polarization losses. The power density curves revealed a substantial difference in peak performance (Fig. 2a). The H₂-PEMFC achieved a PPD exceeding 1000 mW cm⁻², while the PMBI-DMHPFC reached a significantly lower PPD of approximately 600 mW cm⁻², approximately 1.6 times lower. This discrepancy can be explained by several factors inherent to the PMBI-DMHPFC system. Firstly, the electrochemical reactions in PMBI-DMHPFCs, specifically the MOR at the anode, is generally sluggish and more complex than hydrogen oxidation, leading to higher activation losses. In addition, the HPRR involves complex pathways at the cathode and is further complicated by the potential for H₂O₂ decomposition. Secondly, fuel and oxidant crossover reduces overall efficiency. Thirdly, mass transport limitations are exacerbated by CO₂ and O₂ bubbles forming during the electrochemical reactions. While these factors present challenges, advancements in DMFC technology have led to improved performance over time. Fig. 2b provides historical context by showing the

OCV and PPD trend in DMFCs using oxygen O₂ and H₂O₂ as oxidants. The shaded bars represent DMFCs using O₂, while the solid bars represent using H₂O₂. The “This work” data point in Fig. 2b indicates that after optimizing the operating conditions, the PMBI-DMHPFC achieves a significantly higher OCV and PPD than earlier DMFCs. In addition to OCV and PPD, the energy efficiency of PMBI-DMHPFC is higher than that of DMFC (Table S2). The energy consumption of the PMBI-DMHPFC is higher than that of DMFCs (Table S2) due to the non-recirculation of fuel and oxidant solutions, leading to the loss of unused reactants. Introducing a recirculation system would significantly reduce energy consumption, balancing the superior power density and improving the fuel cell's competitiveness despite the additional energy demands of H₂O₂. However, Fig. 2a highlights room to improve PMBI-DMHPFC performance compared to the H₂-PEMFC. These results suggest that while the PMBI-DMHPFC offers a thermodynamic advantage reflected in its high OCV and thermodynamic efficiency (Section S2: thermodynamic efficiency of DMHPFC is $\sim 91\%$ whereas H₂-PEMFC can reach maximum thermodynamic efficiency of $\sim 83\%$), further optimization of MOR and HPRR electrocatalysts, membrane materials to minimize fuel crossover, and cell architectures to improve mass transport are critical to enhance its performance and overall efficiency.

4.2 Effect of anolyte concentration

The concentration of MeOH in the fuel feed is a critical parameter influencing PMBI-DMHPFC performance, balancing fuel supply to the anode with the risks of MeOH crossover and mass transport limitations. To investigate this trade-off, the PMBI-DMHPFC was evaluated with varying MeOH concentrations (1 M to 7 M) while maintaining a fixed KOH concentration of 5 M (Fig. 3a). As shown in Fig. 3a, increasing the MeOH concentration from 1 M to 5 M elevates the MeOH content within the ACL, thereby enhancing MOR kinetics and reducing activation losses. This is evidenced by a slight increase in the

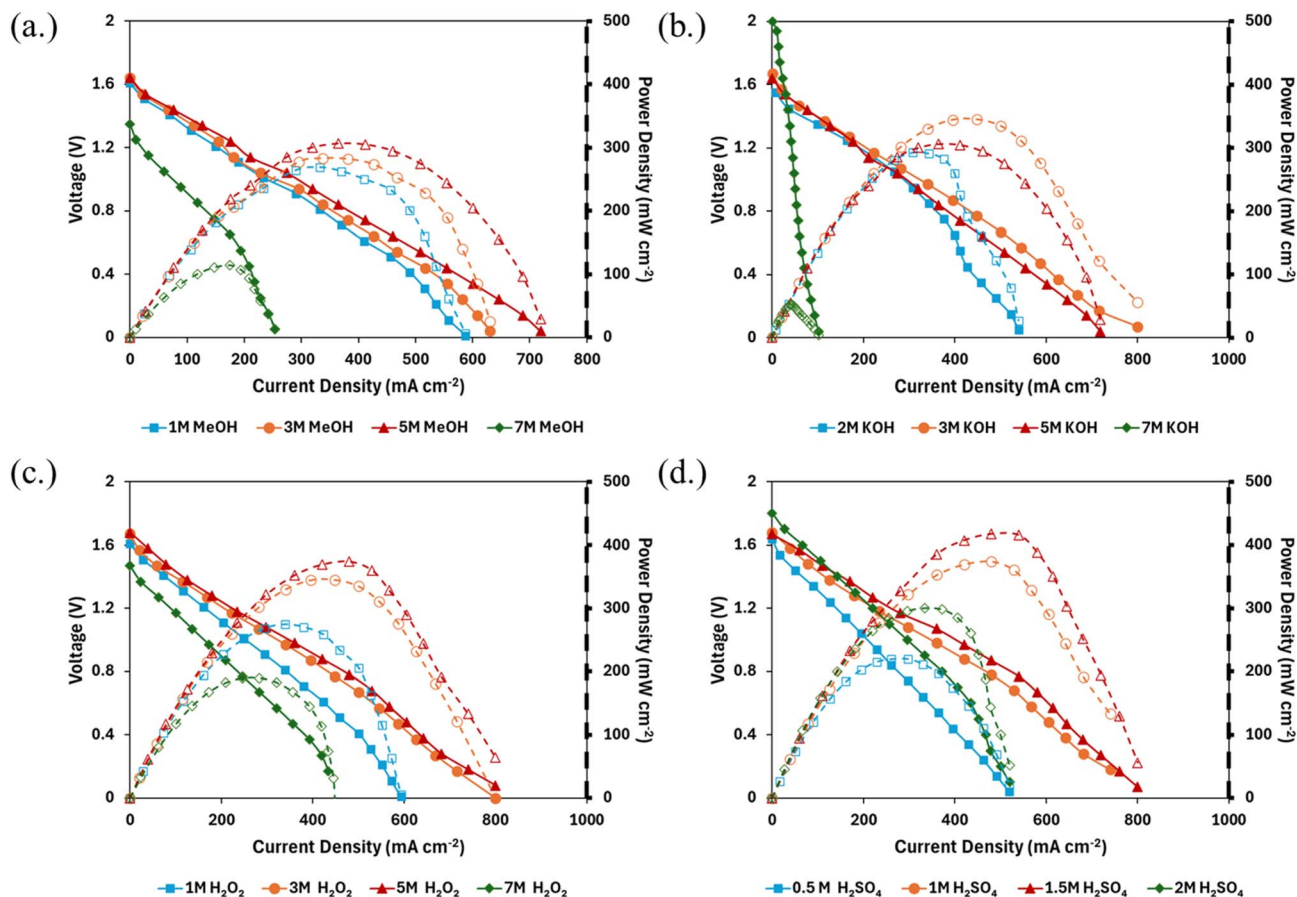


Fig. 3 Polarization and power density curve of the PMBI-DMHPFC with varying (a) MeOH concentration (KOH: 5 M, H_2O_2 : 3 M, H_2SO_4 : 1 M, membrane: N117, flow rate (anode and cathode): $1 \text{ ml min}^{-1} \text{ cm}^{-2}$), (b) KOH concentration (MeOH: 5 M, H_2O_2 : 3 M, H_2SO_4 : 1 M, membrane: N117, flow rate (anode and cathode): $1 \text{ ml min}^{-1} \text{ cm}^{-2}$), (c) H_2O_2 concentration (MeOH: 5 M, KOH: 3 M, H_2SO_4 : 1 M, membrane: N117, flow rate (anode and cathode): $1 \text{ ml min}^{-1} \text{ cm}^{-2}$) and (d.) H_2SO_4 concentration (MeOH: 5 M, KOH: 3 M, H_2O_2 : 5 M, membrane: N117, flow rate (anode and cathode): $1 \text{ ml min}^{-1} \text{ cm}^{-2}$).

OCV from 1.61 V to 1.64 V and an increase in the PPD from 269 to 306 mW cm^{-2} . However, a further increase to 7 M proves detrimental. As depicted in Fig. 3a, at 7 M the OCV decreases significantly to 1.35 V, and the PPD decreases to 114 mW cm^{-2} . This decline suggests that methanol crossover is exacerbated at higher concentrations, leading to cathode catalyst poisoning. Beyond MeOH crossover, a higher methanol concentration also limits mass transport, leading to MeOH's excessive occupation of active sites. This, in turn, causes insufficient OH^- supply, greater OH^- concentration loss, and a decline in cell performance. Therefore, these findings demonstrate that 5 M MeOH provides the optimal balance between MOR kinetics, methanol crossover, and mass transport, leading subsequent experiments to prioritize a 5 M MeOH concentration to maximize power density.

The OH^- concentration also plays a pivotal role in determining the MOR kinetics at the anode. An adequate OH^- concentration is essential for facilitating the MOR, but excessively high concentrations can lead to electrode flooding, hinder mass transport, and cause greater ohmic overpotential. To investigate this effect, the PMBI-DMHPFC was evaluated

with varying KOH concentrations (2 M, 3 M, 5 M, and 7 M) while maintaining a fixed MeOH concentration of 5 M (Fig. 3b). As shown in Fig. 3b, increasing the KOH concentration increased the OCV from 1.55 V to 2.0 V, which is attributed to the thermodynamic favorability of MOR at high OH^- concentrations.¹⁹ Cell voltage increased with increasing KOH concentrations in the low current density region ($<150 \text{ mA cm}^{-2}$). This is attributed to enhanced MOR kinetics due to increased OH^- concentration within the ACL and reduced activation overpotential. In contrast, at higher current densities, the optimal KOH concentration for cell performance was found to be 3 M. Higher KOH concentrations (5 M and 7 M) increased electrolyte viscosity, leading to high ohmic loss and limiting mass transport of reactants (MeOH) and product (CO_2). This is evidenced by the PPD, where the 3 M KOH showed the best results, balancing enhanced MOR kinetics with ohmic resistance and mass transport.

4.3 Effect of catholyte concentration

H_2O_2 concentration is a key parameter in PMBI-DMHPFCs, balancing enhanced HPRR kinetics with increased crossover,



4.4 Effect of membrane thickness

4.5 Effect of flow rate

Fig. 4c shows the polarization curves with different flow rates in the cell. The motivation for increasing the flow rate was to remove CO_2 and O_2 bubbles from the anode and cathode electrodes. CO_2 and O_2 bubbles can increase mass transport losses,

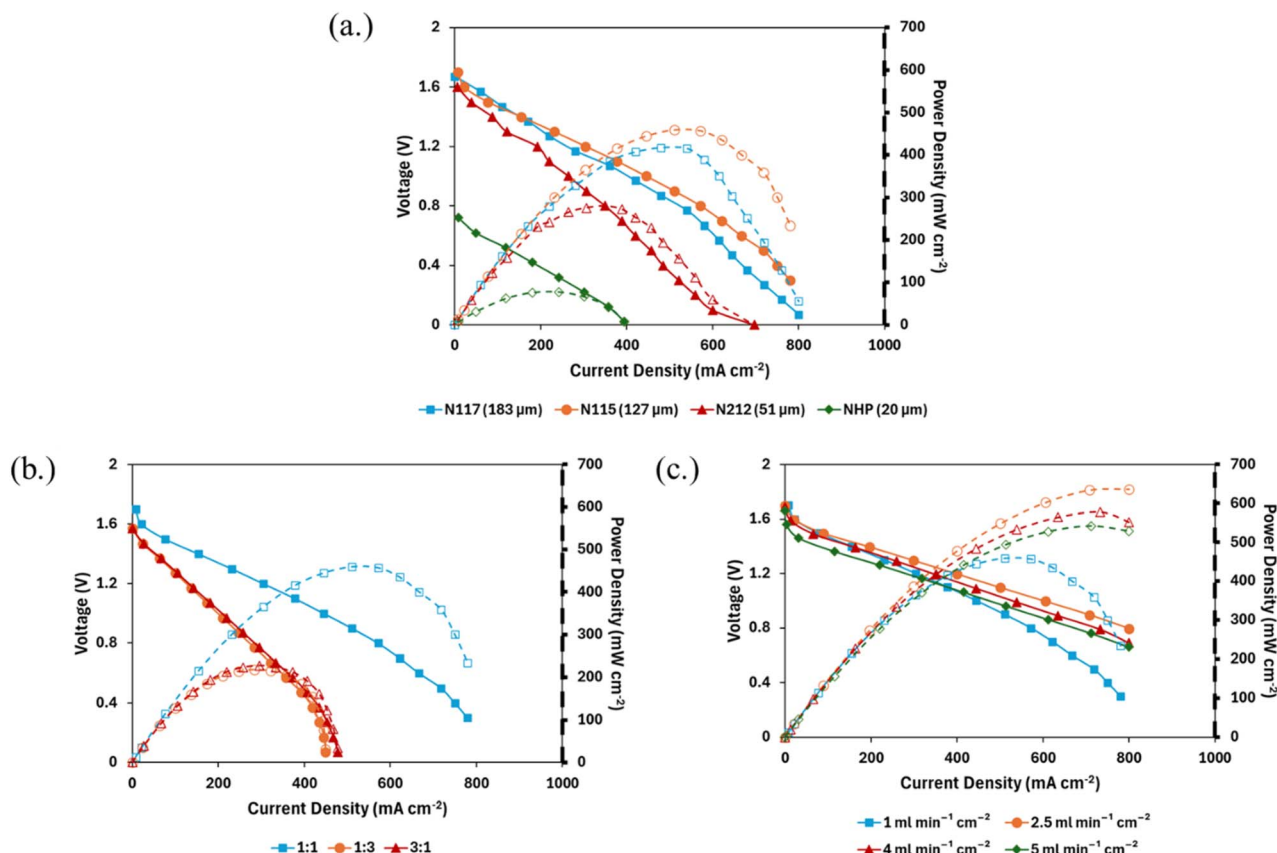


Fig. 4 Polarization and power density curve of the PMBI-DMHPFC with varying (a) membrane thickness (MeOH: 5 M, KOH: 3 M, H_2O_2 : 5 M, H_2SO_4 : 1.5 M, flow rate (anode and cathode): $1 \text{ ml min}^{-1} \text{cm}^{-2}$), (b) flow rate ratio (MeOH: 5 M, KOH: 3 M, H_2O_2 : 5 M, H_2SO_4 : 1.5 M, membrane: N115, base flow rate: $1 \text{ ml min}^{-1} \text{cm}^{-2}$), (c.) flow rate (MeOH: 5 M, KOH: 3 M, H_2O_2 : 5 M, H_2SO_4 : 1.5 M, membrane: N115).

as the bubbles can block the reactant flow and hinder the transport of reactants to the reaction sites. Excessive CO_2 and O_2 accumulation and bubble formation can also damage the PEM, reducing the PMBI-DMHPFC's performance and lifetime. At the cathode, key considerations include limiting the contact duration between the catalyst and the electrolyte to minimize H_2O_2 decomposition, improving reactant utilization by increasing contact duration, and removing O_2 gas bubbles from the catalyst surface to improve HPRR. Hence, increasing electrolyte velocity to remove the adhering bubbles is desirable, but very high velocity will limit the contact time for MOR and HPRR. At high current densities, the $1 \text{ ml min}^{-1} \text{cm}^{-2}$ flow rate's polarization curve exhibited steep voltage drops, indicating significant mass transport limitations due to CO_2 and O_2 bubble accumulation. CO_2 and O_2 bubbles detach from the electrodes as the flow rate increases. However, beyond $2.5 \text{ ml min}^{-1} \text{cm}^{-2}$, the contact duration was limited, hindering both MOR and HPRR. The balance between removing CO_2 and O_2 bubbles and contact duration for reactant utilization led to the optimum flow rate of $2.5 \text{ ml min}^{-1} \text{cm}^{-2}$.

4.6 Constant-current discharge behavior

A constant current discharge test was performed to verify the PMBI-DMHPFC's stability and durability at the identified

optimal operational conditions. A fuel solution containing 5.0 M MeOH and 3.0 M KOH and an oxidant solution containing 5 M hydrogen peroxide and 1.5 M sulfuric acid were fed into the cell at a flow rate of $2.5 \text{ ml min}^{-1} \text{cm}^{-2}$. The operating temperature was 80°C , and the discharge current density was

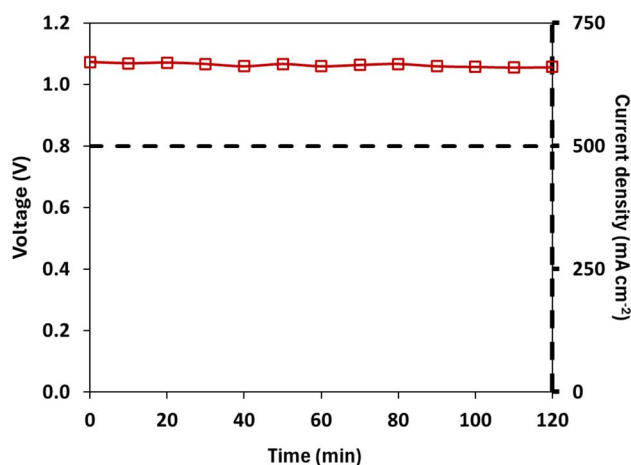


Fig. 5 Constant-current discharging behavior of PMBI-DMHPFC at a current density of 500 mA cm^{-2} (MeOH: 5 M, KOH: 3 M, H_2O_2 : 5 M, H_2SO_4 : 1.5 M, membrane: N115, flow rate: $2.5 \text{ ml min}^{-1} \text{cm}^{-2}$).

5 Conclusion

Conflicts of interest

Data availability

Acknowledgements

References

- Sustainable Energy Fuels, 2025, 9, 5673–5683 | 5681

- 10 B. L. Garcia and J. W. Weidner, Review of Direct Methanol, *Fuel Cells*, 2007, 229–284, DOI: [10.1007/978-0-387-46106-9_5](https://doi.org/10.1007/978-0-387-46106-9_5).
- 11 S. S. Araya, V. Liso, X. Cui, N. Li, J. Zhu, S. L. Sahlin, S. H. Jensen, M. P. Nielsen and S. K. Kær, A Review of The Methanol Economy: The Fuel Cell Route, *Energies*, 2020, 13, 596, DOI: [10.3390/EN13030596](https://doi.org/10.3390/EN13030596).
- 12 C. Lamy, A. Lima, V. LeRhun, F. Delime, C. Coutanceau and J. M. Léger, Recent advances in the development of direct alcohol fuel cells (DAFC), *J. Power Sources*, 2002, 105, 283–296, DOI: [10.1016/S0378-7753\(01\)00954-5](https://doi.org/10.1016/S0378-7753(01)00954-5).
- 13 A. S. Aricò, S. Srinivasan and V. Antonucci, DMFCs: From Fundamental Aspects to Technology Development, *Fuel Cells*, 2001, 1, 133–161, DOI: [10.1002/1615-6854\(200107\)1:2<133::aid-fuce133>3.3.co;2-x](https://doi.org/10.1002/1615-6854(200107)1:2<133::aid-fuce133>3.3.co;2-x).
- 14 S. Sui, X. Wang, X. Zhou, Y. Su, S. Riffat and C. jun Liu, A comprehensive review of Pt electrocatalysts for the oxygen reduction reaction: Nanostructure, activity, mechanism and carbon support in PEM fuel cells, *J. Mater. Chem. A*, 2017, 5, 1808–1825, DOI: [10.1039/C6TA08580F](https://doi.org/10.1039/C6TA08580F).
- 15 L. An, T. Zhao, X. Yan, X. Zhou and P. Tan, The dual role of hydrogen peroxide in fuel cells, *Sci. Bull.*, 2015, 60, 55–64, DOI: [10.1007/s11434-014-0694-7](https://doi.org/10.1007/s11434-014-0694-7).
- 16 R. K. Raman and A. K. Shukla, Electro-reduction of hydrogen peroxide on iron tetramethoxy phenyl porphyrin and lead sulfate electrodes with application in direct borohydride fuel cells, *J. Appl. Electrochem.*, 2005, 35, 1157–1161, DOI: [10.1007/s10800-005-9021-y](https://doi.org/10.1007/s10800-005-9021-y).
- 17 R. K. Raman and A. K. Shukla, A direct borohydride/hydrogen peroxide fuel cell with reduced alkali crossover, *Fuel Cells*, 2007, 7, 225–231, DOI: [10.1002/fuce.200600023](https://doi.org/10.1002/fuce.200600023).
- 18 *Bond Enthalpies*, <https://owl.oit.umass.edu/departments/Chemistry/appendix/bond.html>, accessed July 1, 2024.
- 19 E. H. Yu, K. Scott, R. W. Reeve, L. Yang and R. G. Allen, Characterisation of platinised Ti mesh electrodes using electrochemical methods: Methanol oxidation in sodium hydroxide solutions, *Electrochim. Acta*, 2004, 49, 2443–2452, DOI: [10.1016/j.electacta.2004.01.022](https://doi.org/10.1016/j.electacta.2004.01.022).
- 20 A. I. Dalton and J. V. Bauer, *Method of stabilizing hydrogen peroxide solutions*, 1980.
- 21 M. Pourbaix, *Atlas of Electrochemical Equilibria In-Aqueous Solutions*, 1974.
- 22 M. Ünlü, J. Zhou and P. A. Kohl, Hybrid anion and proton exchange membrane fuel cells, *J. Phys. Chem. C*, 2009, 113, 11416–11423, DOI: [10.1021/JP903252U/ASSET/IMAGES/LARGE/JP-2009-03252U_0008.JPEG](https://doi.org/10.1021/JP903252U/ASSET/IMAGES/LARGE/JP-2009-03252U_0008.JPEG).
- 23 J. Chen, K. Sharma, Z. Wang, S. Sankarasubramanian and V. Ramani, Microscale Bipolar Interfaces for High-Power Fuel Cells, *Acc. Mater. Res.*, 2025, 6(7), 865–875, DOI: [10.1021/ACCOUNTSMR.5C00039/ASSET/IMAGES/LARGE/MR5C00039_0007.JPEG](https://doi.org/10.1021/ACCOUNTSMR.5C00039/ASSET/IMAGES/LARGE/MR5C00039_0007.JPEG).
- 24 Z. Wang, J. Parrondo, C. He, S. Sankarasubramanian and V. Ramani, Efficient pH-gradient-enabled microscale bipolar interfaces in direct borohydride fuel cells, *Nat. Energy*, 2019, 4, 281–289, DOI: [10.1038/s41560-019-0330-5](https://doi.org/10.1038/s41560-019-0330-5).
- 25 Z. Wang, M. Mandal, S. Sankarasubramanian, G. Huang, P. A. Kohl and V. K. Ramani, Influence of Water Transport Across Microscale Bipolar Interfaces on the Performance of Direct Borohydride Fuel Cells, *ACS Appl. Energy Mater.*, 2020, 3(5), 4449–4456, DOI: [10.1021/acsaem.0c00145](https://doi.org/10.1021/acsaem.0c00145).
- 26 Z. Wang, S. Sankarasubramanian and V. Ramani, Reactant-Transport Engineering Approach to High-Power Direct Borohydride Fuel Cells, *Cell Rep. Phys. Sci.*, 2020, 1, 100084, DOI: [10.1016/j.xcrp.2020.100084](https://doi.org/10.1016/j.xcrp.2020.100084).
- 27 D. N. Prater and J. J. Rusek, Energy density of a methanol/hydrogen-peroxide fuel cell, *Appl. Energy*, 2003, 74, 135–140, DOI: [10.1016/S0306-2619\(02\)00139-3](https://doi.org/10.1016/S0306-2619(02)00139-3).
- 28 A. Uddin, L. Dunsmore, H. Zhang, L. Hu, G. Wu and S. Litster, High Power Density Platinum Group Metal-free Cathodes for Polymer Electrolyte Fuel Cells, *ACS Appl. Mater. Interfaces*, 2020, 12, 2216–2224, DOI: [10.1021/ACSAMI.9B13945/ASSET/IMAGES/LARGE/AM9B13945_0008.JPEG](https://doi.org/10.1021/ACSAMI.9B13945/ASSET/IMAGES/LARGE/AM9B13945_0008.JPEG).
- 29 L. Osmieri, R. Escudero-Cid, A. H. A. Monteverde Videla, P. Ocón and S. Specchia, Performance of a Fe-N-C catalyst for the oxygen reduction reaction in direct methanol fuel cell: Cathode formulation optimization and short-term durability, *Appl. Catal., B*, 2017, 201, 253–265, DOI: [10.1016/j.apcatb.2016.08.043](https://doi.org/10.1016/j.apcatb.2016.08.043).
- 30 R. Bashyam and P. Zelenay, A class of non-precious metal composite catalysts for fuel cells, *Nature*, 2006, 443, 63–66, DOI: [10.1038/nature05118](https://doi.org/10.1038/nature05118).
- 31 G. T. K. K. Gunasooriya and M. Saeys, CO Adsorption on Pt(111): From Isolated Molecules to Ordered High-Coverage Structures, *ACS Catal.*, 2018, 8, 10225–10233, DOI: [10.1021/ACSCATAL.8B02371/ASSET/IMAGES/LARGE/CS-2018-02371B_0006.JPEG](https://doi.org/10.1021/ACSCATAL.8B02371/ASSET/IMAGES/LARGE/CS-2018-02371B_0006.JPEG).
- 32 C. Molochas and P. Tsiakaras, Carbon Monoxide Tolerant Pt-Based Electrocatalysts for H₂-PEMFC Applications: Current Progress and Challenges, *Catalysts*, 2021, 11, 1127, DOI: [10.3390/CATAL11091127](https://doi.org/10.3390/CATAL11091127).
- 33 A. H. Ali and P. G. Pickup, Electrolysis of Ethanol and Methanol at PtRu/Pt Catalysts, *J. Electrochem. Soc.*, 2022, 169, 034523.
- 34 M. T. Darby, E. C. H. Sykes, A. Michaelides and M. Stamatakis, Carbon Monoxide Poisoning Resistance and Structural Stability of Single Atom Alloys, *Top. Catal.*, 2018, 61, 428–438, DOI: [10.1007/S11244-017-0882-1/FIGURES/8](https://doi.org/10.1007/S11244-017-0882-1/FIGURES/8).
- 35 S. Y. Huang and C. T. Yeh, Promotion of the electrocatalytic activity of a bimetallic platinum-ruthenium catalyst by repetitive redox treatments for direct methanol fuel cell, *J. Power Sources*, 2010, 195, 2638–2643, DOI: [10.1016/j.jpowsour.2009.11.049](https://doi.org/10.1016/j.jpowsour.2009.11.049).
- 36 O. Sahin and H. Kivrak, A comparative study of electrochemical methods on Pt-Ru DMFC anode catalysts: The effect of Ru addition, *Int. J. Hydrogen Energy*, 2013, 38, 901–909, DOI: [10.1016/j.ijhydene.2012.10.066](https://doi.org/10.1016/j.ijhydene.2012.10.066).
- 37 A. Kormanyos, P. Büttner, M. Bosch, M. Minichova, A. Körner, K. J. Jenewein, A. Hutzler, K. J. J. Mayrhofer, J. Bachmann and S. Cherevko, Stability of Bimetallic Pt_xRu_y – From Model Surfaces to Nanoparticulate Electrocatalysts, *ACS Mater. Au*, 2024, 4, 286–299, DOI: [10.1021/ACSMATERIALSAU.3C00092/ASSET/IMAGES/LARGE/MG3C00092_0006.JPEG](https://doi.org/10.1021/ACSMATERIALSAU.3C00092/ASSET/IMAGES/LARGE/MG3C00092_0006.JPEG).



- 38 S. C. Zignani, V. Baglio, D. Sebastián, T. A. Rocha, E. R. Gonzalez and A. S. Aricò, Investigation of PtNi/C as methanol tolerant electrocatalyst for the oxygen reduction reaction, *J. Electroanal. Chem.*, 2016, **763**, 10–17, DOI: [10.1016/j.jelechem.2015.12.044](https://doi.org/10.1016/j.jelechem.2015.12.044).
- 39 T. Ioroi, K. Yasuda, Z. Siroma, N. Fujiwara and Y. Miyazaki, Enhanced CO-Tolerance of Carbon-Supported Platinum and Molybdenum Oxide Anode Catalyst, *J. Electrochem. Soc.*, 2003, **150**, A1225, DOI: [10.1149/1.1598211/XML](https://doi.org/10.1149/1.1598211/XML).
- 40 Y. W. Zhou, Y. F. Chen, K. Jiang, Z. Liu, Z. J. Mao, W. Y. Zhang, W. F. Lin and W. Bin Cai, Probing the enhanced methanol electrooxidation mechanism on platinum-metal oxide catalyst, *Appl. Catal., B*, 2021, **280**, 119393, DOI: [10.1016/j.apcatb.2020.119393](https://doi.org/10.1016/j.apcatb.2020.119393).
- 41 E. H. Yu, K. Scott, R. W. Reeve, L. Yang and R. G. Allen, Characterisation of platinised Ti mesh electrodes using electrochemical methods: Methanol oxidation in sodium hydroxide solutions, *Electrochim. Acta*, 2004, **49**, 2443–2452, DOI: [10.1016/j.electacta.2004.01.022](https://doi.org/10.1016/j.electacta.2004.01.022).
- 42 R. Serra-Maia, M. Bellier, S. Chastka, K. Tranhuu, A. Subowo, J. D. Rimstidt, P. M. Usov, A. J. Morris and F. M. Michel, Mechanism and Kinetics of Hydrogen Peroxide Decomposition on Platinum Nanocatalysts, *ACS Appl. Mater. Interfaces*, 2018, **10**, 21224–21234, DOI: [10.1021/ACSAMI.8B02345/ASSET/IMAGES/MEDIUM/AM-2018-02345R_M017.GIF](https://doi.org/10.1021/ACSAMI.8B02345/ASSET/IMAGES/MEDIUM/AM-2018-02345R_M017.GIF).
- 43 F. F. Liu and C.-Y. Wang, Mixed Potential in a Direct Methanol Fuel Cell Modeling and Experiments, *J. Electrochem. Soc.*, 2007, **154**, B514.
- 44 X. H. Yan, T. S. Zhao, L. An, G. Zhao and L. Zeng, A novel cathode architecture with a thin reaction layer alleviates mixed potentials and catalyst poisoning in direct methanol fuel cells, *Int. J. Hydrogen Energy*, 2015, **40**, 16540–16546, DOI: [10.1016/j.ijhydene.2015.10.039](https://doi.org/10.1016/j.ijhydene.2015.10.039).
- 45 Z. Wang, J. Parrondo and V. Ramani, Anion Exchange Membranes Based on Polystyrene- Block -Poly(ethylene-ran -butylene)- Block -Polystyrene Triblock Copolymers: Cation Stability and Fuel Cell Performance, *J. Electrochem. Soc.*, 2017, **164**, F1216–F1225, DOI: [10.1149/2.1561712JES/XML](https://doi.org/10.1149/2.1561712JES/XML).

