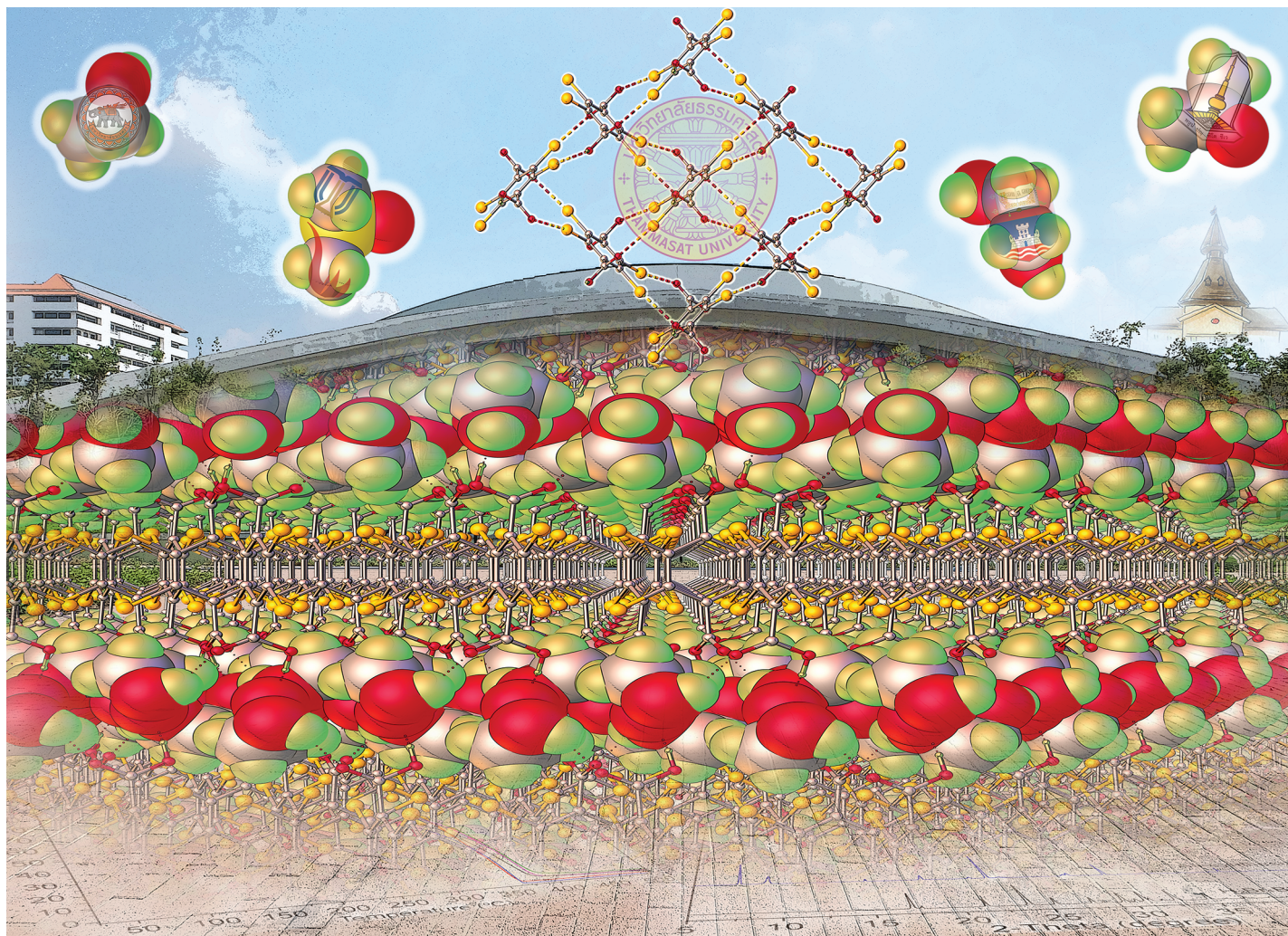




Showcasing collaborative research from the laboratories of Thammasat University, Naresuan University, Mahasarakham University, The Hong Kong University of Science and Technology, and University of Belgrade, led by Dr Kittipong Chainok.

Two-dimensional halogen-bonded organic frameworks based on the tetrabromobenzene-1,4-dicarboxylic acid building molecule

Strong and directional effects of the halogen bonding interactions have been shown to be a useful and reliable crystal engineering tool in the design and construction of robust two-dimensional (2D) halogen-bonded organic frameworks. Such 2D assemblies can function as host layered frameworks and are sufficiently flexible to permit reversible release and adsorption of guest solvent molecules upon the desolvation-resolvation process through the crystal-to-crystal transformation.

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Two-dimensional halogen-bonded organic frameworks based on the tetrabromobenzene-1,4-dicarboxylic acid building molecule†

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Two-dimensional (2D) halogen-bonded organic frameworks were readily engineered by strong and directional effects of the primary Br $\cdots$ O and the secondary Br $\cdots$ π halogen bonding interactions from the tetrabromobenzene-1,4-dicarboxylic acid (H₂Br₄BDC) building molecule involving 100% supramolecular yields. The 2D assembly can function as a host layered framework for the intercalation of various guest solvents including acetone (ATN), ethanol (EtOH), dimethyl sulfoxide (DMSO), and ethylene glycol (EG) resulting in a 1:2 host:guest complexation stoichiometry viz. H₂Br₄BDC·2S (S = ATN (**1**_{ATN}), EtOH (**2**_{EtOH}), DMSO (**3**_{DMSO}), and EG (**4**_{EG})). All the solvates show remarkable similarities in their 2D layered sheets and the bilayer distance significantly responds to the size, shape, molecular conformation, and strength of the hydrogen bonding capability of the intercalated solvent molecules. The transition between solvate formation and desolvation was found to be facile and reversible upon the desolvation–resolvation process. The estimated Br $\cdots$ O halogen bonding energy of the solvates is in the –0.6 to –1.7 kcal mol^{–1} range, which was determined by quantum-mechanical calculations based on density functional theory (DFT) calculations. Furthermore, to quantitatively identify the host–guest intermolecular interactions of these solvates, they were visually compared by Hirshfeld surface analysis.

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Introduction

Noncovalent interactions such as hydrogen bonding, halogen bonding, π – π stacking, and van der Waals force are the fundamental interactions for molecular recognition and self-assembly of discrete molecules to form crystalline solids.¹

These interactions have been widely used in crystal engineering and many aspects of supramolecular chemistry. In particular, the relatively high directionality and strength of hydrogen bonds (up to 40 kJ mol^{–1}) make the prediction and control of molecular orientation in organic solids and discrete coordination complexes feasible.² Over the past few years, hydrogen bonding (HB) has been utilized for the construction of a new class of potentially porous crystalline materials, named hydrogen-bonded organic frameworks (HOFs), through the self-assembly of pure organic building molecules such as guanidinium and sulfonate groups.³ The extended supramolecular framework architectures of HOFs are generally considered as semi-rigid with high flexibility compared to the frameworks created by strong covalent and coordination bonds like covalent organic frameworks (COFs)⁴ and metal–organic frameworks (MOFs).⁵ This is considered as an advantage since guest molecules can be fitted to the highly flexible networks by molecular recognition with supramolecular interactions. This is why research on HOF chemistry has recently been gaining great attention and they are considered as emerging functional materials for a wide range of practical applications such as gas separation,

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chemical sensing, and proton conduction.⁶ In comparison to COFs and MOFs, HOF materials also offer a number of unique features such as easy recrystallization, low density, potentially high thermal stability, and straightforward characterization in both solution and solid state.

Halogen bonding (XB) occurs between a nucleophile and the σ -hole of a polarized halogen atom.⁷ This kind of intermolecular interaction is highly directional and the interaction energies are comparable to HB in which the strength of XB increases in the order $F \ll Cl < Br < I$.⁸ Over the last few decades, XB has appeared as a promising crystal engineering tool in the design and synthesis of new functional supramolecular materials including the so-called halogen-bonded organic frameworks (XBOFs).⁹ In this regard, the most commonly used di-, inter-, and pseudohalogens are known as good halogen bond donors to achieve controlled assembly with halide and aromatic amines as halogen bond acceptors, providing potential applications in ion recognition and light-responsive materials.¹⁰ Related to XB, the so-called halogen- π bond is also an attractive intermolecular interaction between the σ -hole of a highly polarized halogen atom and electron-rich aromatic π -systems.¹¹ The interaction energy of halogen- π interactions is about 50–60% in comparison to the strength of a hydrogen bond.¹² Thus, such interactions are also important and should have a strong effect on the packing and the physical properties of crystalline solids. Intermolecular bonding interactions among molecular organic building blocks in HOFs and XBOFs are much weaker than coordination and covalent bonds in COFs. The control of the organization of the supramolecular arrangement and the accurate prediction of functional properties such as porosities of the resultant supramolecular networks are still very challenging tasks. Furthermore, it is still also very challenging to stabilize HOF and XBOF materials, in which most of the frameworks exhibit an irreversible structural change after the loss of guest molecules or the frameworks collapse during the activations.

On the other hand, the emerging class of two-dimensional (2D) materials such as graphene, silicate clays, and transition metal dichalcogenides has attracted great interest due to their intriguing structural architectures and high surface area to volume ratio, as well as their diverse applications in optoelectronic, energy conversion and storage.¹³ Although the syntheses of HOFs and XBOFs have been well-documented, to date, only a few reported works deal with 2D supramolecular halogen-bonded networks, especially for materials having reversible desolvation/solvation properties.¹⁴ Herein we present novel 2D XBOF assembled materials made from tetrabromobenzene-1,4-dicarboxylic acid (H_2Br_4BDC) building molecules through strong and directional $Br \cdots O$ halogen bonding. The 2D frameworks can act as hosts for the intercalation of various guest solvents including acetone (ATN), ethanol (EtOH), dimethyl sulfoxide (DMSO), and ethylene glycol (EG), resulting in the formation of 1:2 host: guest complexation stoichiometry *viz.* $H_2Br_4BDC \cdot 2S$, where $S = ATN$ (1_{ATN}), EtOH (2_{EtOH}), DMSO (3_{DMSO}), and EG (4_{EG}).

Interestingly, the host frameworks are sufficiently flexible to permit reversible release and adsorption of guest solvent molecules with retention of their crystallinity. All their crystal structures have been determined by single crystal X-ray diffraction analysis at 100(2) and 296(2) K using graphite-monochromatic Cu- $K\alpha$ ($\lambda = 1.5418 \text{ \AA}$) and Mo- $K\alpha$ ($\lambda = 0.7107 \text{ \AA}$) radiation. Their physiochemical properties were also characterized by thermogravimetric analysis (TGA), infrared (FT-IR) spectroscopy, and powder X-ray diffraction (PXRD). Quantum-mechanical calculations based on density functional theory (DFT) calculations were performed for the determination of the energies of halogen bonding. In addition, to quantitatively identify the host-guest intermolecular interactions these solvates were visually compared with the use of Hirshfeld surface analysis.

Experimental

Materials and physical measurement

All chemicals and solvents of analytical grade were purchased from commercial sources and used without further purification. Elemental (C H N) analysis was conducted with a LECO CHNS 932 elemental analyser. IR spectra were recorded on a Perkin-Elmer model Spectrum 100 spectrometer using the ATR mode, in the range of 650–4000 cm^{-1} . Powder X-ray diffraction (PXRD) measurements were carried out on a Bruker D8 ADVANCE X-ray powder diffractometer using Cu $K\alpha$ ($\lambda = 1.5418 \text{ \AA}$) at room temperature. Variable temperature PXRD measurements were collected on a PanAlytical X'Pert PRO diffractometer with a 1D X'celerator detector strip using Cu $K\alpha$ ($\lambda = 1.5418 \text{ \AA}$) with a heating rate of $3 \text{ }^\circ\text{C min}^{-1}$ and measuring a complete diffractogram every $20 \text{ }^\circ\text{C}$ up to $400 \text{ }^\circ\text{C}$. The simulated PXRD patterns were calculated from single crystal X-ray diffraction data and processed by the free Mercury program provided by the Cambridge Crystallographic Data Centre.¹⁵ Thermogravimetric analyses (TGA) were carried out using a Mettler Toledo TGA/DSC3+ from ambient temperature to $500 \text{ }^\circ\text{C}$ with a heating rate of $10 \text{ }^\circ\text{C min}^{-1}$, under a N_2 atmosphere. The solid-state photoluminescence spectra were measured at room temperature using a Horiba Scientific model FluoroMax-4 spectrofluorometer. The solid-state ultraviolet-visible (UV-vis) diffuse reflectance spectra were recorded at room temperature on a U-4100 Spectrometer (Hitachi Corporation, Tokyo, Japan) in the wavelength range of 200–1100 nm.

Crystallization of the solvate forms

$H_2Br_4BDC \cdot 2ATN$ (1_{ATN}). A mixture of H_2Br_4BDC (5 mg) and ATN (2 mL) was added into a 15 mL Teflon lined reactor, sealed in a stainless steel autoclave and placed in an oven. The mixture was heated to $110 \text{ }^\circ\text{C}$ under autogenous pressure for 1 h, and then cooled down to room temperature. Colourless block shaped crystals of 1_{ATN} were obtained. IR ($\nu_{\text{max}}/\text{cm}^{-1}$, s for strong, m medium, w weak): 3376s, 2870m, 2556s, 1750s, 1712s, 1515m, 1084s, 779s.

H₂Br₄BDC·2EtOH (**2**_{EtOH}). The procedure was the same as that used for the synthesis of **1**_{ATN}, except that EtOH was used instead of ATN. Colourless block shaped crystals of **2**_{EtOH} were obtained. IR ($\nu_{\text{max}}/\text{cm}^{-1}$): 3409m, 2986s, 1694s, 1003s, 787s.

H₂Br₄BDC·2DMSO (**3**_{DMSO}). The procedure was the same as that used for the synthesis of **1**_{ATN}, except that DMSO was used instead of ATN. IR ($\nu_{\text{max}}/\text{cm}^{-1}$): 3021s, 1671s, 1024s, 793s.

H₂Br₄BDC·2EG (**4**_{EG}). The procedure was the same as that used for the synthesis of **1**_{ATN}, except that EG was used instead of ATN. IR ($\nu_{\text{max}}/\text{cm}^{-1}$): 3409s, 2878m, 1688m, 1036s, 782s.

Crystallographic data collection and structure refinement

X-ray diffraction data were collected using a Rigaku Oxford-Diffraction SuperNova diffractometer equipped with a multilayer mirror for Cu-K α ($\lambda = 1.54184$ Å) radiation at 100(2) K. Data reduction was performed using CrysAlisPro software¹⁶ and the SCALE3 ABSPACK scaling algorithm was used for absorption correction. The structure was solved with the ShelXT structure solution program using combined Patterson and dual-space recycling methods.¹⁷ The structure was refined by least squares using ShelXL.¹⁸ All non-H atoms were refined anisotropically. The H atoms of solvent molecules were positioned geometrically with C–H = 0.93–0.97 Å and refined using a riding model (AFIX23, AFIX43 and AFIX137 for methine, methylene, and methyl H atoms, respectively in the ShelXL program) with fixed displacement parameters $U_{\text{iso}}(\text{H}) = 1.5U_{\text{eq}}(\text{C})$ for methyl groups and $1.2U_{\text{eq}}(\text{C})$ for the other groups. The O–H hydrogen atoms were located in difference Fourier maps but refined with

O–H = 0.84 ± 0.02 Å. For structure **1**_{ATN}, the solvent acetone molecule was found to be disordered with refined occupancy ratios of 0.493(4):0.507(4). The crystallographic details for the solvates are summarized in Table 1. Crystallographic data for the solvates at 296(2) K are also provided in the ESI† (Table S1). CCDC-1941497–19414504 contain the supplementary crystallographic data for this paper.

Theoretical calculations

All calculations were performed by using the Gaussian 09 (version D.01) program package.¹⁹ All the geometries were extracted from crystal structures type I and type II. For all the calculations of the interaction energies, the ω B97X-D method²⁰ with the def2-TZVPP basis set²¹ was used. Electrostatic potential maps were used to explain the difference between two types of interactions. These maps were calculated from wB97Xd/def2-TZVPP wave functions obtained with Gaussian 09, and visualized with the gOpenMol v3.0 program.²² The surface was defined by the 0.004 a.u. contour of electron density.

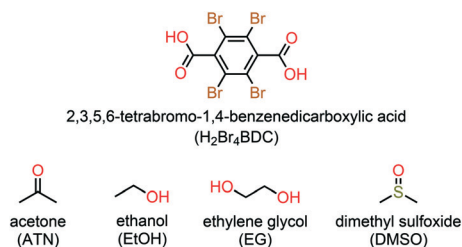
Results and discussion

Synthesis and characterization

Crystallization of H₂Br₄BDC from the polar solvents ATN, EtOH, DMSO, and EG at 110 °C for 1 h in stainless-steel autoclaves with Teflon liners yielded the corresponding solvates of H₂Br₄BDC, Scheme 1. Alternatively, single crystals of these solvates can also be grown by refluxing H₂Br₄BDC in the corresponding solvents at 110 °C for 6 h, filtering the solution, and allowing it to stand undisturbed at ambient temperature for 24 h. However, the crystallization process takes quite a long time compared with the former method.

Table 1 Crystallographic data and refinement summary for all of the solvates at 100(2) K

	1 _{ATN}	2 _{EtOH}	3 _{DMSO}	4 _{EG}
Formula	C ₁₄ H ₁₄ Br ₄ O ₆	C ₁₂ H ₁₄ Br ₄ O ₆	C ₁₂ H ₁₄ Br ₄ O ₆ S ₂	C ₁₂ H ₁₄ Br ₄ O ₈
Formula weight	597.89	573.87	637.99	605.87
Crystal system	Monoclinic	Orthorhombic	Orthorhombic	Orthorhombic
Space group	<i>C2/c</i>	<i>Pnma</i>	<i>Pnma</i>	<i>Pbca</i>
<i>a</i> (Å)	24.9223(4)	9.27849(19)	9.24933(17)	7.96170(10)
<i>b</i> (Å)	9.0196(2)	23.5754(5)	25.7601(6)	9.93120(10)
<i>c</i> (Å)	8.88690(10)	8.6023(2)	8.66218(18)	23.5837(3)
β (°)	93.248(2)	90	90	90
<i>V</i> (Å ³)	1994.47(6)	1881.71(7)	2063.88(7)	1864.75(4)
<i>D</i> _{calc} (g cm ^{−3})	1.991	2.026	2.053	2.158
<i>Z</i>	4	4	4	4
μ (mm ^{−1})	10.086	10.652	11.639	10.880
<i>F</i> (000)	1144	1096	1224	1160
λ (Å)	1.54184 (Cu-K α)	1.54184 (Cu-K α)	1.54184 (Cu-K α)	1.54184 (Cu-K α)
θ range (°)	5.2–73.5	5.4–73.5	5.4–73.5	3.8–73.8
Reflections collected	5459	9416	10 378	9715
Unique reflections	1977	1747	1901	1875
Parameters	155	109	115	121
<i>R</i> _{int} , <i>R</i> _{sigma}	0.016, 0.016	0.045, 0.024	0.034, 0.019	0.022, 0.014
<i>R</i> ₁ , <i>wR</i> ₂ (<i>I</i> > 2 σ (<i>I</i>))	0.017, 0.043	0.029, 0.079	0.045, 0.109	0.019, 0.049
<i>R</i> ₁ , <i>wR</i> ₂ (all data)	0.018, 0.044	0.032, 0.081	0.047, 0.109	0.023, 0.052
GOF on <i>F</i> ² , <i>S</i>	1.096	1.057	1.121	1.058
$\Delta\rho_{\text{max}}$, $\Delta\rho_{\text{min}}$ (e Å ^{−3})	0.37, −0.29	1.01, −0.65	1.71, −1.15	0.36, −0.33



Scheme 1 Molecular structure of the host $\text{H}_2\text{Br}_4\text{BDC}$ molecule and four organic solvent molecules (ATN, EtOH, EG, and DMSO) used in this study.

Attempts to get crystals of $\text{H}_2\text{Br}_4\text{BDC}$ alone by reacting $\text{H}_2\text{Br}_4\text{BDC}$ with either polar or non-polar solvents at different temperatures (such as 40, 60, 80, 100, and 120 °C) were unsuccessful. It should be noted that if the reaction temperature was raised above 120 °C, the $\text{H}_2\text{Br}_4\text{BDC}$ molecules underwent an *in situ* decarboxylation reaction to form tetrabromobenzene. The phase purity of all the solvates was confirmed by matching the PXRD pattern with the simulated pattern obtained based on the SCXRD data (Fig. S1†). FT-IR spectroscopy was also used to investigate the possibility of intermolecular HB and XB interactions between the donor and acceptor in the solvates. The IR spectra of the starting materials and the solvates have been measured and compared (Fig. S2†). The spectra exhibit several significant shifts of a number of key bands. Apparently, the free $\text{H}_2\text{Br}_4\text{BDC}$ shows strong absorption bands at 1725 cm^{-1} and 1390 cm^{-1} , which are characteristic of the asymmetric and symmetric stretching vibrations of the carboxylic groups.²³ These bands are $\sim 15 \text{ cm}^{-1}$ red shifted in all the solvates, suggesting that the simultaneous formation of HB and XB resulted in a reduced electron density on the carbonyl group with respect to the free $\text{H}_2\text{Br}_4\text{BDC}$. The broad bands in the region 3450–3446 cm^{-1} are attributed to the O–H stretching of the carboxylic and hydroxy groups with $\sim 10 \text{ cm}^{-1}$ red shift due to the strong hydrogen bonds involving the hydroxy groups. Notice that the strong C–Br stretching for the free $\text{H}_2\text{Br}_4\text{BDC}$ at 690 cm^{-1} is also red shifted to $\sim 25 \text{ cm}^{-1}$ for all the solvates upon XB formation.

Description of crystal structures

Single crystal X-ray structural analyses revealed that all the solvates form similar centrosymmetric crystals with a 1:2 stoichiometric ratio of the host $\text{H}_2\text{Br}_4\text{BDC}$ and guest solvent molecules. As can be seen in Fig. 1a, the host molecules form a 2D layered framework containing a rectangular grid with dimensions of *ca.* $6.3 \times 6.5 \text{ \AA}^2$. The vacancy of the grids is not large enough to accommodate the guest solvent molecules inside the cavity. Instead, the solvent molecules can be intercalated between two stacks of layers. It should be noted that the carboxyl groups in the single layer are alternatively pointed up and down with respect to the horizontal mean plane defined by the central phenyl rings of the building molecules and they can behave as either hydrogen bond donors or acceptors for the solvent molecules.

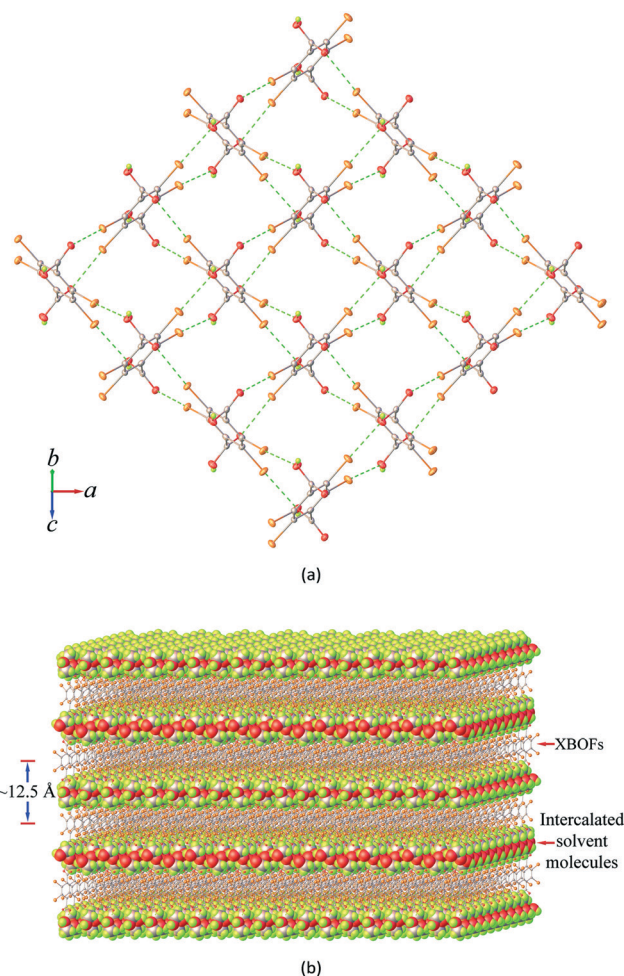


Fig. 1 Perspective view of 1EG as representative showing (a) the 2D halogen-organic frameworks constructed by the $\text{Br}\cdots\text{O}$ halogen bonding interactions (green dashed lines) and (b) the 3D supramolecular structure.

It is very interesting that adjacent $\text{H}_2\text{Br}_4\text{BDC}$ molecules, within the layered framework, are oriented almost perpendicular to and interact with each other through the $\text{Br}\cdots\text{O}$ halogen bonds between the bromine atoms and the oxygen carbonyl or hydroxyl atoms from the carboxyl groups, giving rise to a 2D halogen-bonded framework. Notably, such 2D supramolecular organization is present in all of the four solvates and it involves six $\text{Br}\cdots\text{O}$ contacts per building molecule, thus showing a 100% supramolecular yield. A careful inspection of the structures of these solvates suggests that two types of different supramolecular synthons are observed *viz.* $\text{Br}\cdots\text{O}_{\text{carbonyl}}$ and $\text{Br}\cdots\text{O}_{\text{hydroxyl}}$ (Fig. S3†) and in both cases the distances between the Br and O atoms are closely comparable.

The observed XB interactions are considered strong as indicated by the short $\text{Br}\cdots\text{O}$ distances (2.944(1)–3.114(3) Å), Table 2, in which the experimental values are $\sim 10\%$ shorter than the sum of the Bondi van der Waals radii of the Br and O atoms (3.37 Å),²⁴ and the C–Br $\cdots$ O bond angles (170.2(1)–175.6(1)°) are almost linear. These facts strongly suggest that

Table 2 Parameters of the Br...O halogen bonds in all the solvates

C-Br...O	$d[\text{Br}\cdots\text{O}]$ (Å)		$\angle[\text{C}-\text{Br}\cdots\text{O}]$ (°)	
	100(2) K	296(2) K	100(2) K	296(2) K
1_{ATN}				
C3-Br1...O2 ⁱ	2.944(1)	3.007(2)	174.8(1)	175.6(1)
C4-Br2...O1 ⁱⁱ	3.044(1)	3.132(2)	174.2(1)	175.3(1)
2_{EtOH}				
C3-Br1...O2 ⁱⁱⁱ	2.997(1)	3.062(2)	174.1(1)	174.5(1)
C4-Br2...O1 ^{iv}	2.988(1)	3.093(2)	174.6(1)	175.2(1)
3_{DMSO}				
C3-Br1...O2 ⁱⁱⁱ	3.024(3)	3.098(3)	174.4(1)	175.1(1)
C4-Br2...O1 ^{iv}	2.977(3)	3.050(3)	175.1(1)	175.5(1)
4_{EG}				
C3-Br1...O1 ^v	3.071(1)	3.114(3)	172.9(1)	175.4(1)
C4-Br2...O2 ^{vi}	2.999(1)	3.075(3)	170.2(1)	170.6(1)

Symmetry codes: (i) $x, 1 - y, 0.5 + z$; (ii) $1.5 - x, -0.5 + y, 1.5 - z$; (iii) $-0.5 + x, y, 0.5 - z$; (iv) $0.5 + x, y, 1.5 - z$; (v) $0.5 - x, 0.5 + y, z$; (vi) $-0.5 + x, 1.5 - y, 1 - z$.

the σ -hole of the halogen atom is a key element for the formation of such intermolecular interactions. An interesting feature observed in the structures of these solvates is an apparent auxiliary Br... π contact in which the bromide atom lies above the C-C bond of the phenyl ring. The shortest Br...C contact of about 3.5 Å is slightly larger than the sum of Bondi van der Waals radii, implying weak Br... π interactions^{11a} in these solvates. It should be noted that stronger interactions are known to be less affected by temperature;²⁵ in all cases decreasing temperature down to 100(2) K does not have a strong influence on the Br...O and Br... π distances.

As shown in Fig. 1b, the 2D layers of the host molecules are then held together into a 3D supramolecular structure by host-guest hydrogen bonding. Despite the fact that these solvates show remarkable similarities in their layered frameworks, the geometric features of the hydrogen bonding for host-guest interactions are quite different due to the differences in relative orientation of the solvent molecules confined within the layers. This can be seen from the asymmetric units of each solvate (Fig. S4†), which contain half of a H₂Br₄BDC molecule located on the inversion centre and one complete solvent molecule in a general position. Furthermore, it can be seen that the molecules of the H₂Br₄BDC building unit in these solvates are not planar since the dihedral angles between the carboxylic groups and the tetrabromobenzene moieties are close to 90°. Finally, it is well-known that changes in molecular conformations due to strong and weak intermolecular interactions are very common in molecular host-guest and solvate systems.²⁶ Close inspection of the crystal packing of these solvates reveals that the solvent molecules can function as both hydrogen bond donor and acceptor sites, Fig. 2.

In the case of **1_{ATN}**, the ATN molecule is found to exhibit positional disorder around a pseudo-glide plane symmetry at 296(2) and 100(2) K. The guest solvent molecule participates in strong O-H...O and weak C-H...O hydrogen bonding

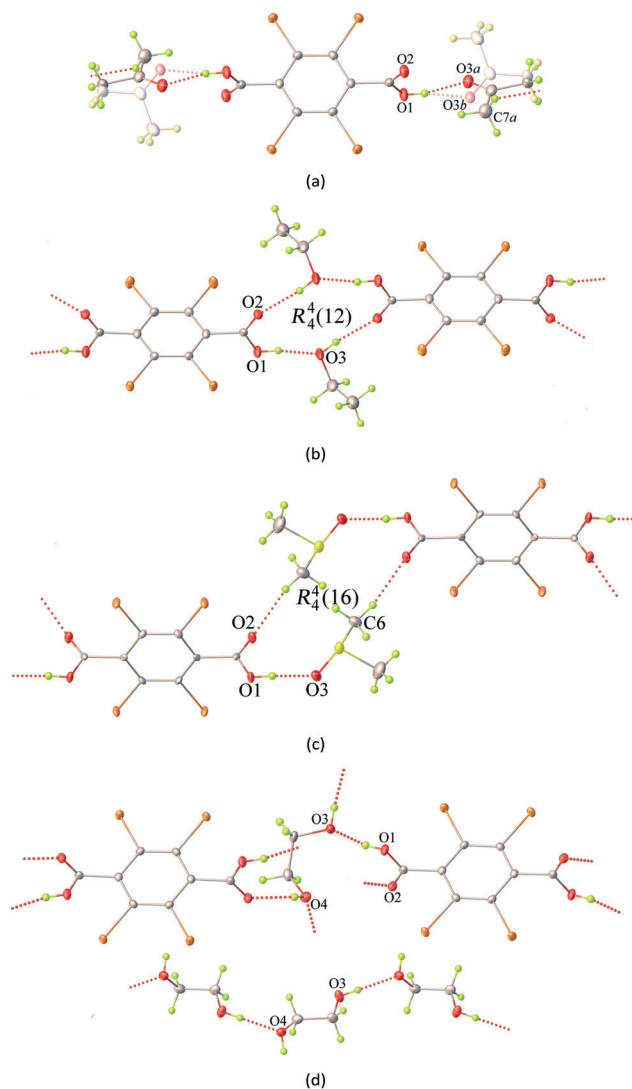


Fig. 2 Detail of the O/C-H...O hydrogen bonding interactions (red dashed-lines) between the H₂Br₄BDC molecules and the solvents for the solvates (a) **1_{ATN}**, (b) **2_{EtOH}**, (c) **3_{DMSO}**, and (d) **4_{EG}**. The head-to-tail O-H...O hydrogen bonding interactions between the EG molecules in **4_{EG}** is also shown (d).

interactions (Fig. 2a), in which the carbonyl and the methyl function as a hydrogen bond acceptor and donor to the respective hydroxy donor and a carbonyl acceptor from the host molecules, giving rise to the R₂²(8) eight-membered ring motif according to the graph-set analysis.²⁷ For **2_{EtOH}**, it is apparent that the orientation of the EtOH molecule is capable of acting as both a hydrogen bond donor and acceptor, which strongly influences the formation of strong O-H...O hydrogen bonds across a centre of inversion. The final supramolecular synthon for **2_{EtOH}** can be described by the R₄⁴(12) graph set twelve-membered ring motif (Fig. 2b). For **3_{DMSO}**, the DMSO solvate molecule can also behave as both a hydrogen bond donor and acceptor, participating in symmetry-related O-H...O and C-H...O hydrogen bonding interactions with carbonyl and hydroxyl oxygen atoms from the host molecules. These interactions give rise to a

supramolecular macrocycle similar to that in the case of 2_{EtOH} and can be described by the $R_4^4(16)$ graph set sixteen-membered ring motif (Fig. 2c). In the case of 4_{EG} , the EG solvate molecule can adopt a *gauche* conformation with respect to the rotation of the central C–C bond and functions as two donors and two acceptors of the hydrogen bonds simultaneously (Fig. 2d). A noteworthy feature of 4_{EG} is that the orientation of the EG molecules facilitates the formation of the head-to-tail O–H $\cdots$ O hydrogen bonds within the layers. It is noted that a carboxyl oxygen atom of the host molecule can act as a bifurcated hydrogen bond and halogen bond acceptor, while a hydroxyl oxygen atom possesses both hydrogen bond donor and halogen bond acceptor properties. As indicated above, stronger intermolecular interactions are known to be less affected by temperature. In this study, lowering the temperatures from 296(2) K to 100(2) K does not lead to any significant differences in the interactions (Table S2†).

Furthermore, a detailed structural inspection of these solvates shows the absence of intermolecular Br $\cdots$ Br halogen bonding and π – π stacking interactions between the phenyl-phenyl aromatic rings. Hence, it is evident that the Br $\cdots$ O/ π halogen bonding and O/C–H $\cdots$ O hydrogen bonding interactions are largely responsible for the observed molecular arrangement and contribute to the stabilization in the formation of these solvates. It should be pointed out that the molecular size, shape, conformation, and hydrogen bonding capability of the solvents could affect the bilayer distances of these solvates. On comparing the distance between adjacent mean planes of the layers defined by the central phenyl rings (100(2) K), the corresponding distances decrease in the order 3_{DMSO} (13.60 Å) > 1_{ATN} (13.12 Å) > 4_{EG} (12.45 Å) > 2_{EtOH} (11.87 Å). Despite this, the total solvent-accessible volume²⁸ of these crystals after removal of guest solvent molecules was estimated to be about 50% per unit cell volume, in which 3_{DMSO} has the largest potential molecular void of 51% (1049 Å³ of the 2064 Å³ unit cell volume).

Hirshfeld surface analysis

Hirshfeld surface analysis and the 2D fingerprint plot are very useful tools for the visualization of intermolecular interactions using different colours and colour intensities, representing short or long contacts and indicating the relative strength of the interactions in crystal structures relative to the van der Waals radii.²⁹ The molecular Hirshfeld surface plots were generated using CrystalExplorer 17 with the 3D d_{norm} surfaces mapped over a fixed colour scale of –0.7606 (red) to 1.7819 (blue) Å. The red and blue colours are used for shorter and longer contacts, respectively, while white highlights for contacts around the van der Waals separations. It should be noted that the bond lengths to hydrogen atoms are set to typical neutron values (C–H = 1.083 Å, O–H = 0.983 Å).³⁰

Hirshfeld surface analysis and the total area of the molecular surface for the $\text{H}_2\text{Br}_4\text{BDC}$ host molecule with all of

the solvates are highly comparable. With the aid of decomposed 2D fingerprint plots (Fig. S5†), the most important contributions to the stabilization of the packing of the host molecules in these solvates come from the Br $\cdots$ H/H $\cdots$ Br, Br $\cdots$ C/C $\cdots$ Br, and Br $\cdots$ O/Br $\cdots$ O contacts (Table S3†). According to the structural analysis results from SCXRD studies in the preceding section, the formation of 2D XBOFs was mainly driven by molecular self-assembly through the Br $\cdots$ O and Br $\cdots$ π halogen bonding interactions involving the host molecules. As shown in Fig. 3a, the dominant interactions between Br and O atoms, corresponding to the Br $\cdots$ O halogen bonds, can be clearly seen in the Hirshfeld surface as the red spots. The relative contributions of these contacts to the total d_{norm} surfaces are 12.7%, 15.4%, 13.8%, and 13.0% for 1_{ATN} , 2_{EtOH} , 3_{DMSO} , and 4_{EG} , respectively, and appear as symmetrical sharp spikes centred near a (d_{e} + d_{i}) sum of ~ 2.9 Å in the 2D fingerprint plots (Fig. S6†), where d_{e} is the distance from a point on the Hirshfeld surface to the nearest external atom and d_{i} is the distance from the same point on the Hirshfeld surface to the nearest atom internal to the surface. The white spots represent the self-

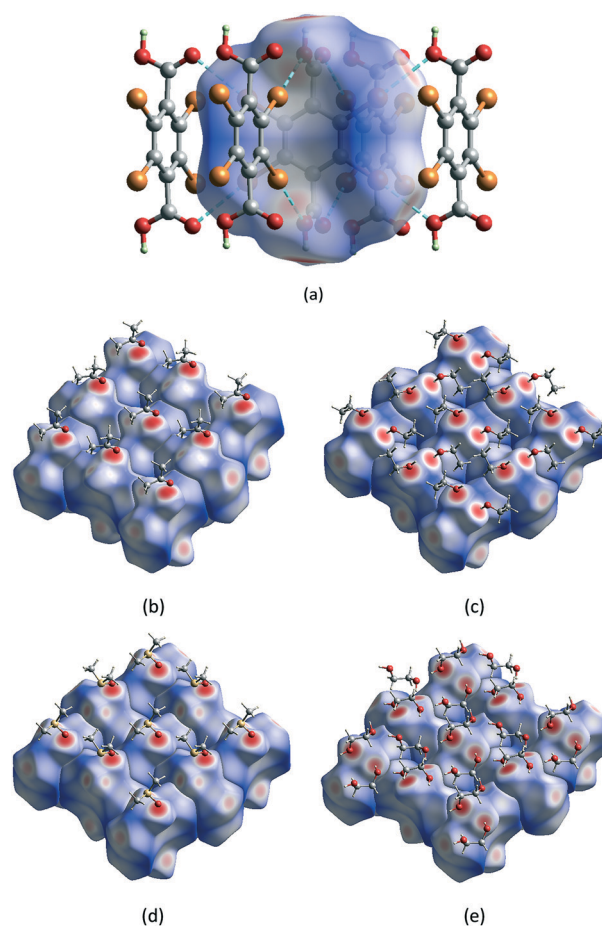


Fig. 3 Hirshfeld surfaces mapped with d_{norm} for (a) the neighbouring host molecules associated with close Br $\cdots$ O contacts shown as red spots and dashed lines, and the intermolecular host-guest hydrogen bonding interactions between the $\text{H}_2\text{Br}_4\text{BDC}$ and solvent molecules for (b) 1_{ATN} , (c) 2_{EtOH} , (d) 3_{DMSO} , and (e) 4_{EG} .

complementary weak $\text{Br}\cdots\text{C}/\text{C}\cdots\text{Br}$ contacts resulting from the $\text{Br}\cdots\pi$ halogen bonding interactions. The proportions of these contacts are comparable in all independent $\text{H}_2\text{Br}_4\text{BDC}$ molecules, comprising $\sim 25\%$ of the contribution with a symmetrical sharp spike in the fingerprint plot having ($d_e + d_i$) ~ 3.4 Å. Other visible spots in the Hirshfeld surfaces correspond to $\text{Br}\cdots\text{H}/\text{H}\cdots\text{Br}$ and $\text{Br}\cdots\text{Br}$ contacts (Table S3†), shown as white areas. These contacts are over 15% longer than the sum of Bondi van der Waals radii. Evidently, it was observed that the contribution from the $\text{C}\cdots\text{C}$ contact on the Hirshfeld surfaces for the host molecules in all the solvates is equal to zero. This confirms the absence of intermolecular $\pi\cdots\pi$ stacking interactions, which is qualitatively consistent with experimental SCXRD analysis. Hence, it can be concluded that the primary $\text{Br}\cdots\text{O}$ and the secondary $\text{Br}\cdots\pi$ halogen bonding interactions play a central role in stabilizing the formation of the supramolecular XBOFs among all the solvates reported here.

Furthermore, the intermolecular host–guest interactions have also been visualized *via* Hirshfeld surface analysis. By selecting the host molecules as the object, the vivid red spots are clearly visible on the d_{norm} surfaces, Fig. 3b–e, attributed to the $\text{H}\cdots\text{O}/\text{O}\cdots\text{H}$ contacts, which correspond to strong (host) $\text{O}\cdots\text{H}\cdots\text{O}$ (guest) hydrogen bonds involving the acidic hydrogen atoms. These contacts are represented by two unsymmetrical narrow pointed areas with a $d_e + d_i$ sum range from ~ 1.5 Å to ~ 1.7 Å (Fig. S7†), meaning that the host molecules can act as hydrogen bond donors and also constitute the acceptor counterpart. On the other hand, by selecting only the solvent molecule as the object, the strong (guest) $\text{O}\cdots\text{H}\cdots\text{O}$ (host) and (guest) $\text{O}\cdots\text{H}\cdots\text{O}$ (guest) interactions are evident on the Hirshfeld surfaces with the contribution of $\text{H}\cdots\text{O}/\text{O}\cdots\text{H}$ contacts varying from 27.7% in 1_{ATN} to 49.6% in 4_{EG} . The contacts appear as wings having a $d_e + d_i$ sum range from ~ 1.5 Å to ~ 2.2 Å in the 2D fingerprint plot (Fig. S8†). There are, however, some differences between these solvents due to the differences in packing motifs and intermolecular arrangements in the solid state. For 1_{ATN} and 2_{EtOH} , the most significant contribution to the d_{norm} surface of the solvents comes from $\text{H}\cdots\text{H}$ contacts, which comprise about half of the total Hirshfeld surfaces. Meanwhile the $\text{H}\cdots\text{O}/\text{O}\cdots\text{H}$ contacts are the main contributors in 3_{DMSO} and 4_{EG} (Table S3†), which suggest that solvent DMSO and EG molecules have stronger intermolecular interaction.

DFT calculations

The interaction energies between Br and O atoms in the crystal structures of these solvates have also been investigated by performing quantum chemical calculations. The type I crystal structure has a network of interactions between the $\text{H}_2\text{Br}_4\text{BDC}$ molecules that consist of two different interactions between these molecules (Fig. S3a†). The first consists of two Br atoms interacting with two O atoms of the carbonyl groups, and the other one is where two Br atoms interact with two O atoms of the hydroxyl groups. Since

attraction between two molecules in crystal structures is not only caused by the interactions of Br and O atoms, but also by the van der Waals attraction, we made model systems to evaluate these two contributions. To get the energy of $\text{Br}\cdots\text{O}$ interactions, we made two model systems, one with molecules as in the crystal structure and a second model system in which we have replaced the interacting COOH group with hydrogen atoms to get the interaction energy between these molecules without the $\text{Br}\cdots\text{O}$ interactions. The difference between these two interaction energies is the interaction energy of Br atoms with the COOH groups. For the $\text{Br}\cdots\text{O}\cdots\text{H}$ contacts, the interaction energy in the model system with COOH groups (Fig. S9a†) is -4.66 kcal mol $^{-1}$, while the interaction energy in the model system without COOH groups (Fig. S9b†) is -3.55 kcal mol $^{-1}$. Hence, the interaction energy of two $\text{Br}\cdots\text{O}\cdots\text{H}$ group contacts is -1.11 ($-4.66 - (-3.55)$) kcal mol $^{-1}$, whereas the interaction energy for one $\text{Br}\cdots\text{O}\cdots\text{H}$ group contact is -0.55 kcal mol $^{-1}$. For the $\text{Br}\cdots\text{O}=\text{C}$ contacts, the interaction energy in the model system with COOH groups (Fig. S10a†) is -6.96 kcal mol $^{-1}$, while the interaction energy in the model system without COOH groups (Fig. S10b†) is -3.55 kcal mol $^{-1}$. Thus, the interaction energy of two $\text{Br}\cdots\text{O}=\text{C}$ contacts is -3.41 ($-6.96 - (-3.55)$) kcal mol $^{-1}$, and the interaction energy for one $\text{Br}\cdots\text{O}=\text{C}$ contact is -1.71 kcal mol $^{-1}$.

The type II crystal structure has a network of $\text{Br}\cdots\text{O}$ interactions that consist of two equivalent contacts between the $\text{H}_2\text{Br}_4\text{BDC}$ molecules; in each contact, there is one interaction of the Br atom with the O atom of the carbonyl group, and a second interaction of the Br atom with the O atom of the hydroxy group (Fig. S3b†). In a similar manner to the type I structure, to investigate the energies of the particular interactions three models were used. One as in the crystal structure and the other two are models in which we replaced an interacting COOH group with a H atom to get the interaction energy between only one Br and O pair. The model system with $\text{Br}\cdots\text{O}=\text{C}$ interaction (Fig. S11a†) has an interaction energy of -5.82 kcal mol $^{-1}$, while the model system without $\text{Br}\cdots\text{O}=\text{C}$ interaction (Fig. S11b†) has an energy of -4.04 kcal mol $^{-1}$. Thus, the $\text{Br}\cdots\text{O}=\text{C}$ interaction energy is the difference, -1.78 kcal mol $^{-1}$. The model system without $\text{Br}\cdots\text{O}\cdots\text{H}$ interaction (Fig. S11c†) has an energy of -5.23 kcal mol $^{-1}$. Therefore, the $\text{Br}\cdots\text{O}\cdots\text{H}$ interaction energy is equal to -0.59 ($-5.82 - (-5.23)$) kcal mol $^{-1}$. The calculated interaction energies of $\text{Br}\cdots\text{O}=\text{C}$ and $\text{Br}\cdots\text{O}\cdots\text{H}$ interactions are very similar in type I and type II structures (Table S4†), and these results also indicate that $\text{Br}\cdots\text{O}=\text{C}$ interactions are significantly stronger.

To investigate the influence of the type of halogen element in these systems, we have calculated the interaction energies of the model systems in which we have replaced the Br atoms with Cl atoms. Model systems based on the type-I crystal structure were used as the starting point and after replacement of Br atoms with Cl atoms, these model systems were optimized at the $\omega\text{B97X-D}/\text{def2-SVPP}$ level of theory. The bond length for the optimized Cl–C bond of 1.73 Å is shorter

than the Br–C bond with 1.89 Å, which is consistent with the difference of their van der Waals radii. Optimized model systems and model systems where COOH groups were replaced with H atoms were used to derive the Cl/O=C and Cl/O–H interaction energies, as was previously done for Br/O=C and Br/O–H interaction energies. Cl/O=C and Cl/O–H interaction energies (–1.05 and –0.43 kcal mol^{–1}, respectively) were less attractive than Br/O=C and Br/O–H interaction energies (–1.71 and –0.55 kcal mol^{–1}, respectively). This was expected since the sigma hole interaction of the Br atom should be stronger than that of the Cl atom, as is well known and can be observed in the electrostatic potential map in Fig. 4. Our attempt to optimize the model system with the Cl halogen atom that is based on the type-II crystal structure was not successful, since the optimized structure had a completely different geometry with different interactions. Hence, we optimized the dimer with hydrogen bonds between COOH groups and parallel planes of the aromatic rings (without Cl/O=C and Cl/O–H interactions). Based on the optimized structure of the H₂Cl₄BDC molecule and Cl/O=C and Cl/O–H distances from previous calculations, we have constructed a reasonable model system based on the type-II crystal structure. This model system and the model systems where the COOH group was replaced with a H atom were used to evaluate the Cl/O=C and Cl/O–H interaction energies, as was previously done for Br/O=C and Br/O–H interactions. The interaction energies were similar to the interaction energies that were previously obtained for Cl/O=C and Cl/O–H interaction energies (–0.98 and –0.32 kcal mol^{–1}, respectively).

Furthermore, the calculated electrostatic potential map of the H₂Br₄BDC and H₂Cl₄BDC molecules can explain the calculated interaction energies. As shown in Fig. 4b, the potential is slightly positive above the ring, and the potential of the Br atoms is negative, with a positive sigma hole in the plane of the ring, opposite to the Br–C bond.³¹ This positive potential forms the electrostatic interaction with negative potential on oxygen atoms of the other molecule. The

electrostatic potential shows significantly stronger negative potential on C=O oxygen than on O–H oxygen, which is in accordance with the calculated interaction energies. The calculated electrostatic potential map of the H₂Cl₄BDC molecule is very similar to the map of the H₂Br₄BDC molecule with the exception of the smaller σ -hole on the Cl atoms than that on the Br atoms. This explains the weaker interaction energies calculated for the H₂Cl₄BDC molecule.

Solvate stability and structural reversibility

To determine the thermal behaviour and to probe the solvate stoichiometry, thermogravimetric (TG) analysis was performed for all the solvates and the H₂Br₄BDC molecule. As can be seen in Fig. 5a, the TG curve of the H₂Br₄BDC building molecule shows that there is no weight loss before the sample was melted (275 °C), confirming that the sample of H₂Br₄BDC did not contain any solvent. Meanwhile the TG curves of the solvates of H₂Br₄BDC merely undergo two steps of weight loss. For all the solvates, the initial weight loss is in the range 50–215 °C and is due to the release of the exclusion of solvate molecules. The second weight loss is attributed to the decomposition of the H₂Br₄BDC molecules. It was observed that the loss of solvent molecules from the solvates follows the order 3_{DMSO} > 4_{EG} > 2_{EtOH} > 1_{ATN}. These trends

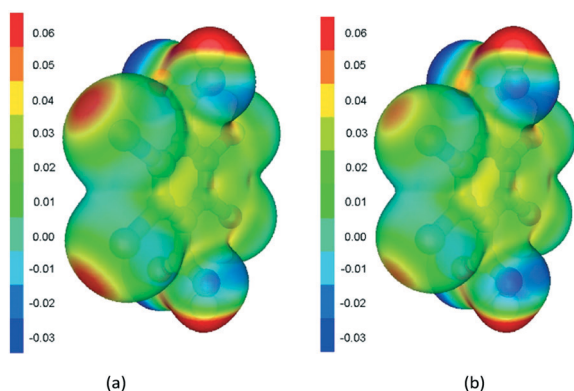


Fig. 4 Electrostatic potentials plotted at the outer contour of an electron density of 0.004 a.u. and values vary from –0.03 to 0.06 a.u. for H₂Br₄BDC (a) and H₂Cl₄BDC (b). All calculations were performed at the ω B97X-D/def2-TZVPP level of theory.

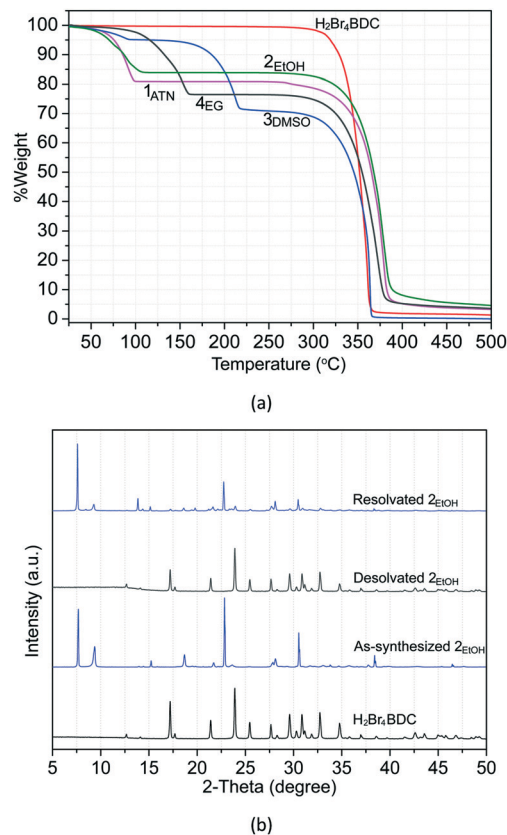


Fig. 5 TG curves of H₂Br₄BDC and all the solvates (a) and PXRD patterns (bottom to top) of H₂Br₄BDC, as synthesized 2_{EtOH}, desolvated 2_{EtOH}, and resolvated 2_{EtOH} (b).

may be attributed to the melting points of the solvents and the appearance of the stronger intermolecular hydrogen bonding interactions as discussed above. It should be noted that the TG curves show no significant difference between the melting points of the solvates and the pure $\text{H}_2\text{Br}_4\text{BDC}$. Furthermore, from the weight loss seen in the TG curves of these solvates the stoichiometry of the $\text{H}_2\text{Br}_4\text{BDC}$:solvent was calculated to be 1:2, which is in reasonably good agreement with the experimental X-ray crystallographic analysis.

Moreover, an interesting feature is that structural reversibility was observed through the desolvation–resolution process. As observed above, the TG results indicate that all the solvates show similar thermal stability in which all the solvates convert to the desolvated forms before melting. Thus, the study of desolvation–resolution of the ethanol solvate 2_{EtOH} is described in detail as a representative example. When the ethanol molecules were removed by heating at 40 °C under vacuum (~ 10 mbar pressure) for 1 h, the solvate form indeed changes to the $\text{H}_2\text{Br}_4\text{BDC}$ phase. The original solvate form can be easily recovered by adding the ethanol solvent into the desolvated phase and subsequent heating at 110 °C for 1 h in an autoclave. As can be seen from the PXRD patterns in Fig. 5b, the positions of the diffraction peaks for the desolvated and reabsorbed solvent phases are perfectly in accordance with those in original $\text{H}_2\text{Br}_4\text{BDC}$ and as-synthesized 2_{EtOH} , respectively, showing that the crystal structure remains essentially unchanged in each case. More interestingly, this reversibility of the desolvation–resolution process for this solvate 2_{EtOH} could be repeated at least ten times. In addition, variable temperature PXRD was also carried out to monitor the structural changes after the solvent leaves (Fig. S12†). It was verified that after removal of lattice solvent molecules at ~ 100 °C, the molecules rearranged to the original $\text{H}_2\text{Br}_4\text{BDC}$, which was followed by the decomposition of the sample at ~ 250 °C.

Photochemical properties

In addition, the photochemical properties of the synthesized solvates, as well as the parent $\text{H}_2\text{Br}_4\text{BDC}$ molecule were investigated. Both UV-vis and photoluminescence spectra were recorded for the samples (Fig. S13†). The UV-vis spectra of all of the compounds show strong absorption below 320 nm. Inspection of this part of the spectrum reveals slight differences. Compounds $\text{H}_2\text{Br}_4\text{BDC}$, 1_{ATN} , and 2_{EtOH} exhibit a trough around 300 nm and shoulders around 245 nm. On the other hand, the spectra of 3_{DMSO} and 4_{EG} have a minimum in this part at 230 nm followed by a gradual increase towards the peak at 300 nm. Furthermore, a small shoulder is observed in these spectra at 358 nm. This feature is most prominent in the spectrum of 4_{EG} . Upon excitation at 300 nm, weak emission spectra can be observed for these compounds. Once again, the spectra of $\text{H}_2\text{Br}_4\text{BDC}$, 1_{ATN} , and 2_{EtOH} are somewhat similar, exhibiting fairly broad peaks between 320 and 390 nm. The spectra of the solvates are slightly red shifted and structured in comparison to the

starting material. In contrast the emission spectra of 3_{DMSO} and 4_{EG} are narrower and blue shifted in comparison to the previous ones. Meanwhile the spectrum of 4_{EG} exhibits a distinct peak at 325 nm; the peak for 3_{DMSO} merges with the tail of the scattered excitation light. The spectral differences observed in both methods, especially for the solvate samples, are likely to be a reflection of the different hydrogen bond strengths discussed above.

Conclusions

In summary, halogen bonding has been shown to be a useful and reliable crystal engineering tool for the formation of solvate crystals. In this work, the construction of 2D halogen-bonded organic frameworks in the crystalline state was readily controlled through the strong and directional $\text{Br}\cdots\text{O}$ halogen bonding interactions (-0.6 to -1.7 kcal mol $^{-1}$) between $\text{H}_2\text{Br}_4\text{BDC}$ molecules, involving 100% supramolecular yields. These 2D assemblies are considered as host frameworks; meanwhile the carboxylic groups of the $\text{H}_2\text{Br}_4\text{BDC}$ building molecules with both hydrogen bond donor and acceptor sites can also form hydrogen bonding with various guest solvents (*i.e.* ATN, EtOH, DMSO, and EG), and subsequently solvates with 1:2 host:guest stoichiometry are formed. These solvates show interesting facile and reversible crystalline-to-crystalline transformation upon a desolvation–resolution process and this reversible process of such transformation could be repeated at least ten times as monitored by PXRD. Such studies give some insight into the requirements for the elaboration of layered frameworks, which is necessary in order to ultimately engineer such an emerging class of 2D supramolecular materials for the adsorption of solvent organic molecules. Currently, we are investigating the effects of different solvents and anions on the XB interactions in such systems.

Conflicts of interest

There are no conflicts to declare.

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Notes and references

- (a) *Crystal Design: Structure and Function*, ed. G. R. Desiraju, Wiley, New York, 2003; (b) *Frontiers in Crystal Engineering*, ed. E. R. T. Tiekink and J. J. Vittal, John Wiley & Sons, Ltd, Sussex, 2006.

- 2 (a) C. B. Aakeröy and K. R. Seddon, *Chem. Soc. Rev.*, 1993, **22**, 397–407; (b) G. R. Desiraju, *Acc. Chem. Res.*, 2002, **35**, 565–573; (c) T. Adachi and M. D. Ward, *Acc. Chem. Res.*, 2014, **47**, 2514–2524.
- 3 (a) J. Luo, J.-W. Wang, J.-H. Zhang, S. Lai and D.-C. Zhong, *CrystEngComm*, 2018, **20**, 5884–5898; (b) R.-B. Lin, Y. He, P. Li, H. Wang, W. Zhou and B. Chen, *Chem. Soc. Rev.*, 2019, **48**, 1362–1389.
- 4 (a) A. P. Côté, A. I. Benin, N. W. Ockwig, A. J. Matzger, M. O’Keeffe and O. M. Yaghi, *Science*, 2005, **310**, 1166–1170; (b) S. Yuan, X. Li, J. Zhu, G. Zhang, P. V. Puyvelde and B. V. D. Bruggen, *Chem. Soc. Rev.*, 2019, **48**, 2665–2681.
- 5 (a) S. S.-Y. Chui, S. M.-F. Lo, J. P. H. Charmant, A. G. Orpen and I. D. Williams, *Science*, 1999, **283**, 1148–1150; (b) H. Li, M. Eddaoudi, M. O’Keeffe and O. M. Yaghi, *Nature*, 1999, **402**, 276–279; (c) O. M. Yaghi, M. O’Keeffe, N. W. Ockwig, H. K. Chae, M. Eddaoudi and J. Kim, *Nature*, 2003, **423**, 705–714.
- 6 (a) Y. He, S. Xiang and B. Chen, *J. Am. Chem. Soc.*, 2011, **133**, 14570–14573; (b) A. Karmakar, R. Illathvalappil, B. Anothumakkool, A. Sen, P. Samanta, A. V. Desai, S. Kurungot and S. K. Ghosh, *Angew. Chem., Int. Ed.*, 2016, **55**, 10667–10671; (c) D.-D. Zhou, Y.-T. Xu, R.-B. Lin, Z.-W. Mo, W.-X. Zhang and J.-P. Zhang, *Chem. Commun.*, 2016, **52**, 4991–4994; (d) I. Hisaki, Y. Suzuki, E. Gomez, Q. Ji, N. Tohnai, T. Nakamura and T. Nakamura, *J. Am. Chem. Soc.*, 2019, **141**, 2111–2121; (e) X.-T. He, Y.-H. Luo, D.-L. Hong, F.-H. Chen, Z.-Y. Zheng, C. Wang, J.-Y. Wang, C. Chen and B.-W. Sun, *ACS Appl. Nano Mater.*, 2019, **2**, 2437–2445.
- 7 (a) P. Metrangolo, H. Neukirch, T. Pilati and G. Resnati, *Acc. Chem. Res.*, 2005, **38**, 386–395; (b) C. B. Aakeröy, M. Baldrighi, J. Desper, P. Metrangolo and G. Resnati, *Chem. – Eur. J.*, 2013, **19**, 16240–16247; (c) P. Metrangolo and G. Resnati, *Chem. Commun.*, 2013, **49**, 1783–1785; (d) A. Mukherjee, S. Tothadi and G. R. Desiraju, *Acc. Chem. Res.*, 2014, **47**, 2514–2524; (e) L. C. Gilday, S. W. Robinson, T. A. Barendt, M. J. Langton, B. R. Mullaney and P. D. Beer, *Chem. Rev.*, 2015, **115**, 7118–7195; (f) D. Bulfield and S. M. Huber, *Chem. – Eur. J.*, 2016, **22**, 14434–14450.
- 8 (a) T. Brinck, J. S. Murray and P. Politzer, *Int. J. Quantum Chem.*, 1992, **44**, 57–64; (b) A. C. Legon, *Angew. Chem., Int. Ed.*, 1999, **38**, 2686–2714; (c) L. C. Gilday, S. W. Robinson, T. A. Barendt, M. J. Langton, B. R. Mullaney and P. D. Beer, *Chem. Rev.*, 2015, **115**, 7118–7195; (d) G. Cavallo, P. Metrangolo, R. Milani, T. Pilati, A. Priimagi, G. Resnati and G. Terraneo, *Chem. Rev.*, 2016, **116**, 2478–2601.
- 9 (a) V. I. Nikolayenko, D. C. Castell, D. P. van Heerden and L. J. Barbour, *Angew. Chem., Int. Ed.*, 2018, **57**, 12086–12091; (b) S. Shankar, O. Chovnik, L. J. W. Shimon, M. Lahav and M. E. van der Boom, *Cryst. Growth Des.*, 2018, **18**, 1967–1977.
- 10 (a) G. Cavallo, P. Metrangolo, T. Pilati, G. Resnati, M. Sansotera and G. Terraneo, *Chem. Soc. Rev.*, 2010, **39**, 3772–3783; (b) A. Brown and P. D. Beer, *Chem. Commun.*, 2016, **52**, 8645–8658.
- 11 (a) A. N. M. M. Rahman, R. Bishop, D. C. Craig and M. L. Scudder, *CrystEngComm*, 2003, **5**, 422–428; (b) H. Matter, M. Nazar and S. Güssregen, *The importance of π -interactions in crystal engineering*, ed. E. R. T. Tiekink and J. Zukerman-Schpector, Wiley, Chichester, 2012, pp. 187–132; (c) H. Zhuo, Q. Li, W. Lia and J. Cheng, *Phys. Chem. Chem. Phys.*, 2014, **16**, 159–165; (d) M. Giese, M. Albrecht and K. Rissanen, *Chem. Commun.*, 2016, **52**, 1778–1795.
- 12 (a) H. G. Wallnoefer, T. Fox, K. R. Liedl and C. S. Tautermann, *Phys. Chem. Chem. Phys.*, 2010, **12**, 14941–14949; (b) K. E. Riley, C. L. Ford Jr. and K. Demouchet, *Chem. Phys. Lett.*, 2015, **621**, 165–170; (c) A. Bauzá and A. Frontera, *Theor. Chem. Acc.*, 2017, **136**, 1–8; (d) M. Bujak, H.-G. Stammel, S. Blomeyer and N. W. Mitzel, *Chem. Commun.*, 2019, **55**, 175–178.
- 13 (a) F. Ghahari, D. Walkup, C. Gutiérrez, J. F. Rodríguez-Nieva, Y. Zhao, J. Wyrick, F. D. Natterer, W. G. Cullen, K. Watanabe, T. Taniguchi, L. S. Levitov, N. B. Zhitenev and J. A. Strosio, *Science*, 2017, **356**, 845–849; (b) S. Mukherjee and G. Singh, *ACS Appl. Energy Mater.*, 2019, **22**, 932–955.
- 14 (a) P. Metrangolo, Y. Carcenac, M. Lahtinen, T. Pilati, K. Rissanen, A. Vij and G. Resnati, *Science*, 2009, **323**, 1461–1464; (b) J. Martí-Rujas, L. Colombo, J. Lü, A. Dey, G. Terraneo, P. Metrangolo, T. Pilati and G. Resnati, *Chem. Commun.*, 2012, **48**, 8207–8209.
- 15 C. F. Macrae, I. J. Bruno, J. A. Chisholm, P. R. Edgington, P. McCabe, E. Pidcock, L. Rodríguez-Monge, R. Taylor, J. van de Streek and P. A. Wood, *J. Appl. Crystallogr.*, 2008, **41**, 466–470.
- 16 *CrysAlis CCD and CrysAlis RED*, version 1.171.37.35, Oxford Diffraction Ltd, Yarnton, Oxfordshire, U.K., 2014.
- 17 G. M. Sheldrick, *Acta Crystallogr., Sect. A: Found. Adv.*, 2015, **71**, 3–8.
- 18 G. M. Sheldrick, *Acta Crystallogr., Sect. C: Struct. Chem.*, 2015, **71**, 3–8.
- 19 M. J. Frisch, G. W. Trucks, H. B. Schlegel, G. E. Scuseria, M. A. Robb, J. R. Cheeseman, G. Scalmani, V. Barone, G. A. Petersson, H. Nakatsuji, X. Li, M. Caricato, A. Marenich, J. Bloino, B. G. Janesko, R. Gomperts, B. Mennucci, H. P. Hratchian, J. V. Ortiz, A. F. Izmaylov, J. L. Sonnenberg, D. Williams-Young, F. Ding, F. Lipparini, F. Egidi, J. Goings, B. Peng, A. Petrone, T. Henderson, D. Ranasinghe, V. G. Zakrzewski, J. Gao, N. Rega, G. Zheng, W. Liang, M. Hada, M. Ehara, K. Toyota, R. Fukuda, J. Hasegawa, M. Ishida, T. Nakajima, Y. Honda, O. Kitao, H. Nakai, T. Vreven, K. Throssell, J. A. Montgomery Jr., J. E. Peralta, F. Ogliaro, M. Bearpark, J. J. Heyd, E. Brothers, K. N. Kudin, V. N. Staroverov, T. Keith, R. Kobayashi, J. Normand, K. Raghavachari, A. Rendell, J. C. Burant, S. S. Iyengar, J. Tomasi, M. Cossi, J. M. Millam, M. Klene, C. Adamo, R. Cammi, J. W. Ochterski, R. L. Martin, K. Morokuma, O. Farkas, J. B. Foresman and D. J. Fox, *Gaussian 09, Revis. D.01*, 2016.

- 20 J.-D. Chai and M. Head-Gordon, *Phys. Chem. Chem. Phys.*, 2008, **10**, 6615–6620.
- 21 F. Weigend and R. Ahlrichs, *Phys. Chem. Chem. Phys.*, 2005, **7**, 3297–3305.
- 22 D. L. Bergman, L. Laaksonen and A. Laaksonen, *J. Mol. Graphics Modell.*, 1997, **15**, 301–306.
- 23 (a) U. H. F. Bunz, *Chem. Rev.*, 2000, **100**, 1605–1644; (b) P.-Y. Du, W. Gu and X. Liu, *CrystEngComm*, 2016, **18**, 5140–5148.
- 24 A. Bondi, *J. Phys. Chem.*, 1964, **68**, 441–451.
- 25 B. K. Saha, *J. Indian Inst. Sci.*, 2017, **97**, 177–191.
- 26 (a) F.-G. Klärner and B. Kahlert, *Acc. Chem. Res.*, 2003, **36**, 919–932; (b) J. Wang, Y. Gao, J. Xiang, M. Wang and A. Wu, *CrystEngComm*, 2015, **17**, 2245–2249.
- 27 M. C. Etter and J. C. MacDonal, *Acta Crystallogr., Sect. B: Struct. Sci.*, 1990, **46**, 256–262.
- 28 A. L. Spek, *Acta Crystallogr., Sect. C: Struct. Chem.*, 2015, **71**, 9–18.
- 29 (a) M. A. Spackman and J. J. McKinnon, *CrystEngComm*, 2002, **4**, 378–392; (b) J. J. McKinnon, D. Jayatilaka and M. A. Spackman, *Chem. Commun.*, 2007, 3814–3816; (c) M. A. Spackman and D. Jayatilaka, *CrystEngComm*, 2009, **11**, 19–32.
- 30 F. H. Allen, O. Kennard, D. G. Watson, L. Brammer, A. G. Orpen and R. Taylor, *J. Chem. Soc., Perkin Trans. 2*, 1987, S1–S19.
- 31 T. Clark, M. Hennemann, J. S. Murray and P. Politzer, *J. Mol. Model.*, 2007, **13**, 291–296.