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Modulating the porosity of N-doped carbon materials for enhanced CO₂ capture and methane uptake†

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N-doped carbons with modulated porosity have been prepared *via* the addition of melamine or urea as a N source to an activation mixture containing biomass-derived carbonaceous matter of low O/C ratio (air-carbonised date seed, *Phoenix dactylifera*, ACDS), and potassium hydroxide (KOH) as an activating agent. To access a broad range of surface area and mix of porosity characteristics, a series of carbons were prepared by varying the following: (i) the amount of melamine or urea (at melamine or urea/ACDS ratio of 1 or 2), (ii) the KOH/ACDS ratio (2 or 4), and (iii) activation temperature (600, 700, or 800 °C). We found that the N added to the activation mix acts both as an N-dopant and porogen, with the latter effect enabling formation of larger pores, which extended the pore size distribution of resulting porous carbons into the mesopore region. Furthermore, the presence of N acts to increase the surface area and provides carbons with tuneable porosity (with respect to the mix of microporosity and mesoporosity) and variable packing density, all of which may be tailored towards suitability for enhanced uptake of CO₂ and/or methane. The activated carbons have significant N content of up to 18 wt%, surface area of up to 3600 m² g⁻¹, and pore volume that reaches 2.1 cm³ g⁻¹. Depending on the preparation conditions and resulting mix of micro/mesoporosity, the carbons show excellent low pressure CO₂ capture at 25 °C of 1.7 mmol g⁻¹ at 0.15 bar and 4.7 mmol g⁻¹ at 1 bar, and uptake of up to 25 mmol g⁻¹ at 20 bar. The porosity and packing density may also be directed towards excellent methane storage with gravimetric uptake of up to 0.42 g g⁻¹ at 25 °C and 100 bar, volumetric storage capacity of up to 266 cm³ (STP) cm⁻³ at 25 °C and 100 bar, and a working capacity (for 100 to 5 bar pressure swing) of 196 cm³ (STP) cm⁻³.

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1. Introduction

Increasing environmental pollution, caused in large part by excessive use of fossil fuels, is a serious threat to future generations of mankind, and to all forms of life on Earth. The CO₂ produced by the combustion of fossil fuels is widely believed to be a significant factor amongst greenhouse gas (GHG) emissions in causing consequences that engender global warming.^{1,2} It is, therefore, critical to capture CO₂ emissions whilst simultaneously reducing global net GHG emissions *via* consumption reduction and the use of 'green' energy sources. Carbon capture and storage (CCS) *via* the use of solid adsorbents is currently a promising strategy that has attracted significant attention as a viable solution for mitigating the ever-increasing CO₂ emissions.²⁻⁶ Research into alternative energy sources is

encouraged more than ever, with the goal of shifting from fossil fuels to using renewable, cleaner, and sustainable fuels. As an alternative fuel, natural gas (NG), the main component of which is methane, is more environmentally-friendly compared to traditional fossil fuels.⁷⁻¹¹ However, despite the potential benefits of using natural gas, the current storage methods, whether by liquefying or compression, are expensive, energy-intensive, and pose safety concerns, and therefore remain a barrier to its widespread use.⁷⁻¹⁴ Recently, adsorbed natural gas (ANG), which typically involves physisorption onto porous adsorbents, has attracted attention as a potential alternative method for storing NG under ambient temperature and low pressure conditions that would increase the viability of NG. Storing ANG on a solid material has several advantages over traditional liquefied natural gas (LNG) or compressed natural gas (CNG) techniques, including higher energy density, better safety, and improved energy efficiency.^{7-11,15-18} Current research is therefore focused on preparing suitable adsorbents that have the desired properties for high performance methane storage.

Among the current known adsorbent materials, porous carbons have gained considerable attention in gas capture and storage for environmental remediation.¹⁹⁻²⁵ Due to their widespread availability from low-cost and sustainable resources,

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ease of synthesis, high thermal and chemical stability, and tuneable structural properties, activated carbons have potential not only for CCS^{26–35} but also as methane storage materials.^{36–39} Biomass sources, usually in waste form, are among the most commonly used precursors in the production of activated carbon, due to their wide availability and affordability, and carbon-rich nature.^{39–44} The surface area and overall porosity of any adsorbent are important factors by which the total amount of gas adsorbed can be determined. The elemental composition, the nature of the precursors, the mode of carbonisation, and the activation parameters, including the type and amount of activating agent, and the temperature, all have an influence on the structural properties of activated carbons.^{25,36–38} Among the various chemical activation agents, KOH is widely used due to its effectiveness in controlling the surface area and porosity of the resulting carbon *via* adjustment of the activation conditions.^{43,45–47} Carbons obtained *via* this method can exhibit high surface area ($>2800 \text{ m}^2 \text{ g}^{-1}$), and their pore size can be tuned between ultramicropores/micropores and small mesopores.^{19,48} However, although these carbons have shown advantages in certain applications, especially at low-pressure adsorption, their use in energy-related applications at high-pressures remains limited, due to limits on their mesoporosity.^{49,50} Only a few examples of precursors, such as polypyrrole,⁵⁰ imidazolium-based ionic liquids,⁵¹ zeolite-templated carbons,⁵² or graphene,⁵³ have generated activated materials with a high level of mesoporosity when activated with KOH.

The surface area and porosity of activated carbons can be modified further by adding so-called mediators.^{48,54,55} For example, Fuertes and Sevilla prepared high-surface area carbons *via* the KOH-activation of hydrochar, using melamine as a mediator.⁵⁶ While a large surface area is important for high-pressure gas storage, the materials must exhibit other characteristics to be suitable for methane uptake. For instance, when materials have a high surface area and pore volume, the packing density decreases, causing a decrease in volumetric gas uptake.⁵⁷ Hence, in order to obtain a material with suitable properties for methane adsorption, the adsorbent should exhibit an optimal balance between possessing a high surface area, mix of microporosity/mesoporosity, and a high packing density.

We have recently demonstrated that the atomic ratio of oxygen to carbon (O/C ratio) of any precursor (carbonaceous matter) is a key factor *via* which the porosity and packing density can be predicted and modulated.^{35–38,43,45,46} Depending on the type of biomass and the mode of carbonisation, the behaviour of carbonaceous matter during activation can vary.^{34–36,38,43} For example, carbonaceous matter with a high O/C ratio generates activated carbons with high surface area and a large pore volume but with a low packing density.^{25,58,59} In contrast, a precursor with a low O/C ratio generates carbons with a high degree of microporosity and a high packing density, but with a surface area that hardly exceeds $2500 \text{ m}^2 \text{ g}^{-1}$.^{25,35,37,38,43,46} This phenomenon is attributed to the susceptibility, or resistance, of a precursor to activation.^{25,36–38,43} As a consequence, the porosity, and in particular the micropore/mesopore mix, can be influenced by the O/C ratio of the

precursor, which in turn plays a key role in predicting the packing density, which is a key factor in determining volumetric gas uptake.^{24,34,48,60–62} An activated carbon with a high packing density is highly desirable for methane uptake, especially when coupled with high surface area.

The objective of this work was to synthesize activated carbons with modulated porosity, appropriate for CO₂ and/or methane storage, by adding a mediator. Specifically an N-containing additive was added to the precursor during activation. In an attempt to enhance both the porosity and packing density, a precursor with a low O/C ratio, namely air-carbonised date seeds (*Phoenix dactylifera*), designated as ACDS, which is known to be resistant to activation, was utilised.³⁸ The low O content of the ACDS carbon was expected to favour generation of activated carbons with a high packing density, while the presence of N-containing additives would work to increase the surface area. This combination is expected to generate activated carbons with high surface area, tuneable porosity (micropore/mesopore mix), and variable packing density. The interplay between these three factors was expected to yield carbons with a range of desirable properties suited for CO₂ and/or methane storage.

2. Experimental section

2.1 Synthesis of N-doped carbons

2.1.1 Air carbonisation. Date seeds (*Phoenix dactylifera*) were used as the starting material for this study. Before use, the date seeds were washed with deionised water and dried in an oven at 80 °C. 5 g of the washed date seeds in an alumina boat were placed in a horizontal tube furnace, which was then heated to 400 °C at a heating ramp rate of 10 °C min⁻¹ under 100 cm³ min⁻¹ flow of nitrogen gas. Once at 400 °C, the date seeds were exposed to a flow of air for 5 min., and then the furnace was allowed to cool down under a flow of nitrogen gas. The carbonaceous material obtained was designated as ACDS (air-carbonised date seeds).

2.1.2 Chemical activation. The ACDS carbon was chemically activated with KOH in the presence of melamine as a N source. In a typical procedure, the ACDS carbon was thoroughly mixed in an agate mortar with appropriate amounts of KOH (at KOH/ACDS weight ratio of 2 or 4) and melamine (at melamine/ACDS weight ratio of 1 or 2). The resulting mixture was placed in an alumina boat and heated in a horizontal furnace to the desired temperature (600, 700 or 800 °C) at a heating ramp rate of 3 °C min⁻¹ under a nitrogen gas flow and held at the target temperature for 1 h, following which the samples were allowed to cool down to room temperature under a nitrogen atmosphere. The resulting activated carbons were then thoroughly washed by stirring in an aqueous solution of HCl (10%) at room temperature to remove any residual inorganic salts. Note, precautions should be taken when washing a nitrogen-containing precursor and potassium salts, as hazardous amounts of toxic potassium cyanate (KOCN) or potassium cyanide (KCN) could be produced. The carbons were then repeatedly washed with deionised water until neutral pH (pH ~ 7) was achieved for the filtrate, and then dried at 100 °C in an



oven. The activated carbons are designated as DSM xT - y , where DSM stands for the combination of date seeds (DS) and melamine (M), x is the KOH/ACDS weight ratio, T is the activation temperatures (600, 700 or 800 °C), and y is the melamine/ACDS weight ratio. For comparison purposes, two activated carbons were prepared in the presence of urea (at urea/ACDS weight ratio of 1 or 2) at KOH/ACDS ratio of 4 and 800 °C. These activated carbons were designated as DSU4800-1 or DSU4800-2, where U indicates use of urea, 4 is the KOH/ACDS ratio, 800 is the activation temperature (800 °C), and 1 or 2 are the urea/ACDS ratio. It is important to note that activation of melamine and urea was attempted and that they yielded no activated carbons (ESI Fig. S1†).

2.2 Materials characterisation

An exeter analytical CE-440 elemental analyser was used for CHN analysis. A TA instruments SDT Q600 analyser was used to perform thermogravimetric analysis (TGA) in ambient air (100 mL min⁻¹) conditions at heating ramp rate of 10 °C min⁻¹. Powder XRD analysis was performed using a PANalytical X'Pert PRO diffractometer with a Cu-K α light source (40 kV, 40 mA), a step size of 0.021, and step time of 50 s. Porosity analysis and textural properties of the activated carbons were determined using nitrogen sorption analysis (at -196 °C) using a Micromeritics 3FLEX sorptometer. The samples were degassed under vacuum for 16 h at 240 °C prior to analysis. The surface area was calculated using the Brunauer–Emmett–Teller (BET) method applied to nitrogen sorption isotherms according to Rouquerol rules, in the relative pressure (P/P_0) range of 0.02–0.22. The relative pressure range used to calculate surface area was maintained so that the multipoint BET fitting produced a positive y -axis intercept (*i.e.*, $C > 0$) and $V_{\text{ads}}(1 - P/P_0)$ increased with P/P_0 . The total pore volume was determined from the total nitrogen adsorbed at a relative pressure near saturation ($P/P_0 \approx 0.99$). The micropore surface area and micropore volume were obtained *via* t -plot analysis. The pore size distributions (PSDs) were determined by applying the non-local density functional theory (NL-DFT) method to the nitrogen adsorption data with the assumption of a slit pore model. Scanning electron microscopy (SEM) images were obtained using a JEOL 7000F FEG-SEM microscope. Transmission electron microscopy (TEM) images were obtained using a JEOL 2100F instrument operating at 200 kV and equipped with a Gatan Orius CCD for imaging. Prior to analysis, the carbon samples were suspended in distilled water and dispersed onto lacey carbon support films.

2.3 CO₂ and methane uptake measurements

Using a Hiden Isochema intelligent gravimetric analyser (IGA-003), CO₂ uptake of the carbons was determined in the pressure range of 0–20 bar at room temperature. The methane uptake was determined using a Hiden Isochema XEMIS analyser at 25 °C and a pressure range of 0–100 bar. In all cases, the carbons were degassed at 240 °C for several hours before performing the gas uptake measurements.

3. Results and discussion

3.1 Yield and elemental composition

The yields and/or elemental composition of the raw date seeds, air carbonised date seeds (ACDS), and activated carbons are summarised in Tables 1 and 2. The raw date seeds have a high C content and relatively low O content with an O/C ratio of ~ 0.66 , which is relatively low for biomass where typical values are between 0.75 and 1.0.³⁸ When biomass is subjected to air-carbonisation, the O/C ratio of the resulting carbonaceous matter is reduced to a greater extent than for other carbonisation methods such as hydrothermal carbonisation or conventional pyrolysis.⁴³ For this reason we adopted air-carbonisation to convert the raw date seeds to carbonaceous matter. Following air-carbonisation, the C content increased from 49.4 wt% for the raw date seeds to 76.7 wt% for the ACDS carbon. On the other hand, the apparent O content decreased from 43.5 wt% to 17.5 wt%, while the H content decreased from 6 wt% to 3.8 wt%. Thus the O/C ratio of the ACDS carbon is 0.171, which is quite low for biomass-derived carbonaceous matter.³⁸ For the premixed precursors (DSM- x and DSU- x), the nominal N content increases significantly while the amount of C and O reduce compared to the ACDS carbon. For activated carbons prepared at melamine/ACDS ratio of 1, the N content is between 0.8 and 8.4 wt% depending on the severity of activation. At melamine/ACDS ratio of 2, the activated carbons have N content of between 1.2 and 18.0 wt%, the latter of which is remarkably high. The N content of the activated carbons decreases with rise in activation temperature and/or amount of KOH used. For example, the N content of DSM2600-1 is 8.4 wt%, which reduces to 2.5 wt% for DSM2800-1, and 4.6 wt% for DSM4600-1. In this way, the N/C ratio decreases at higher activation temperature (ESI Fig. S2†).

In general, the C content of the activated carbons increases compared to that of the precursors, which is accompanied by a decrease in the amount of N, H and O, except that some samples activated at the lowest temperature show a decrease in their C content associated with their very high N content. Thus,

Table 1 Yield and/or elemental composition of raw date seeds, air-carbonised date seeds (ACDS) and activated carbons prepared at melamine or urea/ACDS ratio of 1

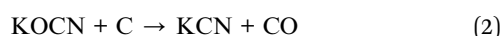
Sample	Yield [wt%]	C [%]	H [%]	N [%]	O [%]	(O/C) ^a	(N/C) ^a
Raw DS	—	49.4	6.0	1.1	43.5	0.660	0.019
ACDS	50	76.7	3.8	2.0	17.5	0.171	0.022
DSM-1	—	51.5	4.0	36.0	8.5	0.124	0.599
DSU-1	—	46.1	5.3	26.2	22.4	0.364	0.487
DSM2600-1	40	73.4	1.5	8.4	16.7	0.171	0.098
DSM2700-1	31	78.6	0.9	7.2	13.3	0.127	0.079
DSM2800-1	25	86.9	0.1	2.5	10.5	0.091	0.025
DSM4600-1	32	74.8	0.5	4.6	20.1	0.202	0.053
DSM4700-1	28	78.0	0.3	3.8	17.9	0.172	0.042
DSM4800-1	16	90.0	0.0	0.8	9.2	0.077	0.008
DSU4800-1	21	92.6	0.1	0.5	6.8	0.055	0.005

^a Atomic ratio



the C content of activated carbons derived from DSM-1 increased from 73.4 wt% for DSM2600-1 to 90.0 wt% for DSM4800-1, while the H and O content decreased from 1.5 and 16.7 wt% to nil and 9.2 wt%, respectively. For DSM-2-derived carbons, the C content increases from 67.7 wt% (DSM2600-2) to 89.6 wt% (DSM4800-2), while the H content decreases from 1.5 to 0.2 wt% and the O content lowers from 12.8 to 9.2 wt%.

The flash air-carbonisation of raw date seeds resulted in a 50% yield of ACDS carbon. For the carbons obtained from DSM-1, the yield is between 16 and 40 wt% and for those derived from DSM-2, the yield is in the range of 10 to 33 wt%. The lower yield at high activation temperature may be ascribed to the decomposition of endogenic carbon nitride. The presence of melamine increased the susceptibility of the ACDS carbon to activation, meaning that the mechanism of activation is likely different from that of N-free precursors.^{48,55,63–65} At elevated temperatures, the K⁺ salts etch N species, resulting in N loss from the surface by the formation of inorganic salts (KOCN or KCN) according to eqn (1) and (2).^{64,65} The strength of interaction between N and K⁺ depends on how much of each is present in the precursor. This is consistent with the elemental composition of the activated carbons and shows that the overall loss of N, especially for high temperature activation, is related to the K/melamine ratio (N : K⁺) during the activation process.



Altogether, it seems that K⁺ attacks surface N species easily and preferentially, even when there are large amounts of surface O.⁶⁴ This provides further evidence that the activation mechanism involves a surface-mediated etching reaction, which has the effect of decreasing the overall yield of the carbons.

3.2 Thermal stability and morphology

Thermogravimetric analysis (TGA) was performed to ascertain the purity of the carbons and their thermal stability. After an initial weight loss below 100 °C, which is attributed to the evaporation of adsorbed moisture, the activated carbons are stable up to 500 °C (ESI Fig. S3 and S4†). This is followed by a significant weight loss in a single step as a result of carbon combustion. In general, the carbons have hardly any residual mass, indicating that they are carbonaceous materials with minimal, trace or nil amounts of inorganic content. X-ray diffraction (XRD) patterns (ESI Fig. S5 and S6†) show broad features at 2-theta = 24.7° and 43° assigned to (002) and (100) diffractions of graphitic carbon, respectively. The intensity of these broad peaks reduced when the amount of activating agent was increased (ESI Fig. S6†), which supports the expectation that the K⁺ preferentially attacks N atoms, which leads to greater amorphisation. SEM images of representative carbons prepared at low KOH/ACDS ratio reveal a conchoidal morphology (Supporting Fig. S7†), while carbons prepared at higher KOH/ACDS have rough-surface morphology. HRTEM images (Supporting Fig. S8†) reveal a disordered porous structure, which is typical for activated carbons.

3.3 Porosity and textural properties

The nitrogen sorption isotherms and corresponding pore size distribution (PSD) curves of activated carbons prepared at melamine or urea/ACDS ratio of 1 are shown in Fig. 1 and 2. The isotherm type depends on the amount of KOH and/or the activation temperature. Thus, as shown in Fig. 1, carbons obtained at a KOH/ACDS ratio of 2 exhibit type I isotherms indicative of microporosity but with a broader adsorption “knee” for samples prepared at higher activation temperature (*e.g.*, DSM2800-1). The broadening in the adsorption knee indicates the presence of larger pores (supermicropores and small mesopores), as confirmed by the PSD curves in Fig. 1B. The isotherms of samples prepared at KOH/ACDS ratio of 4 are either type I (DSM4600-1 and DSM4700-1) or show a greater proportion of mesoporosity at higher activation temperature (DSM4800-1) as shown in Fig. 2. The isotherms of DSM4600-1 and DSM4700-1 are primarily type I but with a broader adsorption knee, indicating the presence of larger micropores. For activation at 800 °C (DSM4800-1 and DSU4800-1), the isotherm shape was close to type IV. The near mesoporous nature of DSM4800-1 and DSU4800-1 contrasts with the microporous nature of carbons prepared from activation of ACDS alone.³⁷ Clearly, the presence of melamine and urea, which introduce N to the activation mix, has the effect of increasing the susceptibility of the ACDS carbon to activation thus generating larger pores with dimensions in the large micropore and small mesopore range as shown in Fig. 2B.

The nitrogen sorption isotherms and PSD curves of activated carbons prepared at melamine or urea/ACSD ratio of 2 are shown in Fig. 3 and 4. In general, except for samples generated at a high severity of activation (DSM4800-2 and DSU4800-2), all samples show a type I isotherm, typical of microporous materials. The total amount of nitrogen adsorbed rises with the activation temperature and the amount of KOH used. Carbons prepared at a KOH/ACDS ratio of 2 (Fig. 3) are microporous although with a broad adsorption knee for DSM2800-2. For KOH/ACDS ratio of 4 (Fig. 4), the isotherms for DSM4600-2 and DSM4700-2 are typically microporous while those for DSM4800-2 and DSU4800-2 shift towards significant mesoporosity (*i.e.*, type IV character). Overall, the isotherms of samples prepared at melamine/ACDS ratio of 2 are very similar to those generated at a ratio of 1, although the latter suggest higher levels of porosity. This indicates that use of excessive amounts of N-dopants, *i.e.*, melamine or urea/ACDS ratio beyond 1, does not significantly alter the porosity.

The porosity depicted in Fig. 1 and 4 clearly shows that the N : K ratio plays an important role in pore formation. For DSMxT-1 carbons, porosity levels range from microporous to mesoporous depending on the amount of KOH and the activation temperature. For samples DSM2600-1, DSM2700-1 and DSM2800-1, the porosity falls in between micro/supermicroporous to small mesoporous (1.0–2.5 nm pores), with moderate N content retained in the carbons. For samples activated with higher amounts of KOH (KOH/precursor ratio of 4), there was a greater loss of N accompanied by the formation of larger pores. Bimodal pore size distributions are seen in these



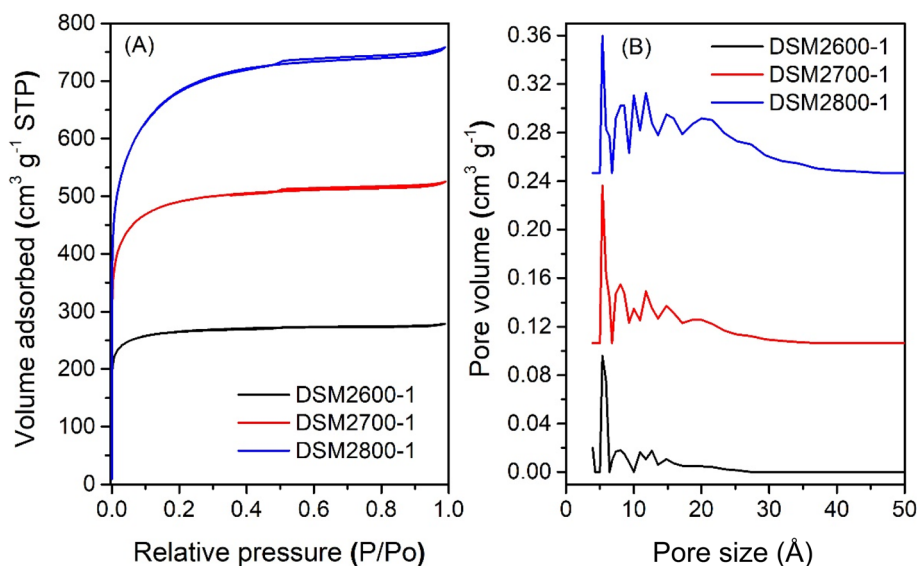


Fig. 1 Nitrogen sorption isotherms (A) and pore size distribution curves (B) of activated carbons prepared at melamine/ACDS ratio of 1 and KOH/ACDS ratio of 2.

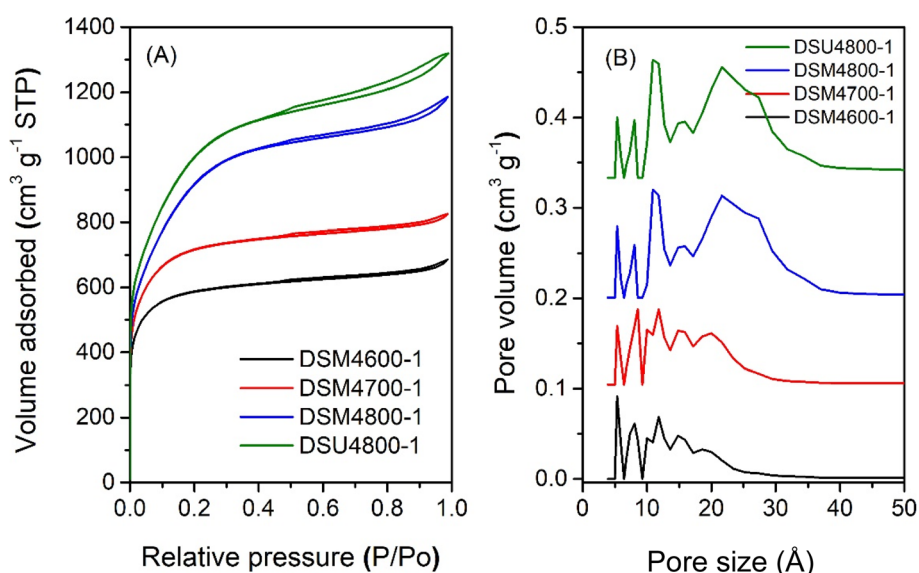


Fig. 2 Nitrogen sorption isotherms (A) and pore size distribution curves (B) of activated carbons prepared at melamine or urea/ACDS ratio of 1 and KOH/ACDS ratio of 4.

samples, with a smaller percentage of micropores centred at *ca.* 0.8 nm and 1.1 nm, and a much greater proportion of mesopores centred at *ca.* 2.5 nm, as shown in Fig. 1B. Thus the proportion of mesopores in DSM2T-1 carbons is lower than for DSM4T-1 samples, which may be ascribed to greater formation of K₂CO₃ leading to more etching out of N species with the overall effect of increasing porosity.

On the other hand, at melamine/ACDS ratio of 2, there is an increase in the N content relative to the K in the mixture (*i.e.*, rise in N/K ratio), meaning an excess amount of N, which is then retained in the activated carbon at amounts of up to 18 wt% as shown in Table 2. A higher amount of N can also lead to the

formation of KCN, an unreacted species, which can result in a decrease in porosity at apparently high N content.⁶⁴ Thus DSM2600-2, DSM2700-2, and DSM2800-2, have ultra-microporous characteristics with pores centred at 0.5 to 1.1 nm and hardly any pores larger than 2 nm. At melamine/ACDS ratio of 2 and higher KOH/precursor ratio of 4, lowering in the N : K ratio resulted in the removal of more N (samples DSM4T-2). For activation at 600 and 700 °C, samples DSM4600-2 and DSM4700-2 are micro- and super-microporous and dominated by 1.2 nm pores and some small mesopores. For activation at 800 °C (DSM4800-2 and DSU4800-2), the PSD is comparable to DSM4800-1 and DSU4800-1, with mesopores



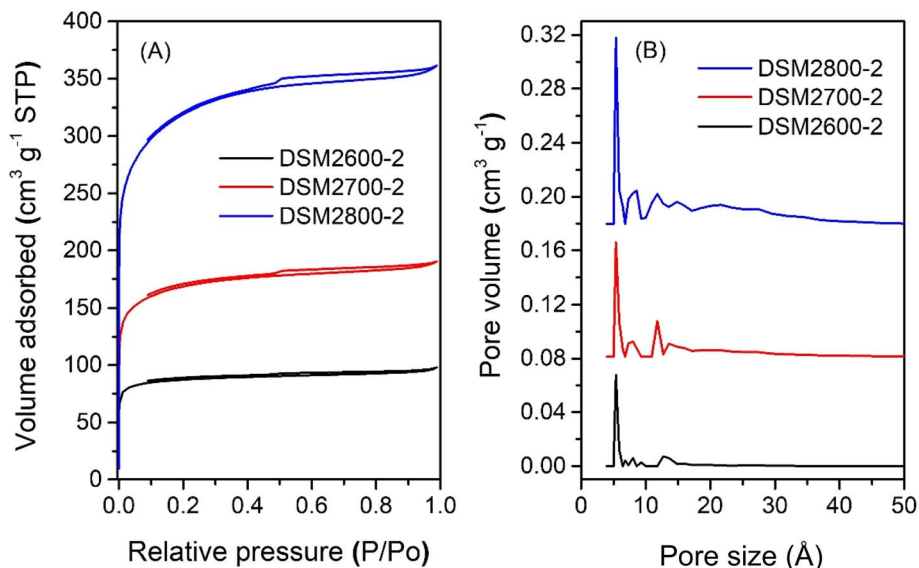


Fig. 3 Nitrogen sorption isotherms (A) and pore size distribution curves (B) of activated carbons prepared at melamine/ACDS ratio of 2 and KOH/ACDS ratio of 2.

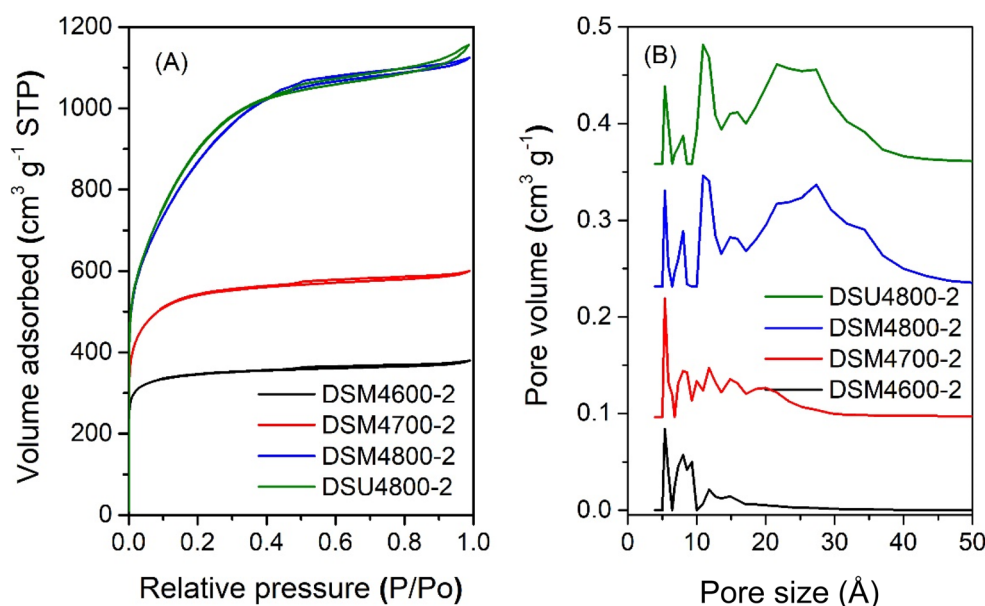


Fig. 4 Nitrogen sorption isotherms (A) and pore size distribution curves (B) of activated carbons prepared at melamine or urea/ACDS ratio of 2 and KOH/ACDS ratio of 4.

centred at 2.7 nm. It appears that when higher amounts of melamine are used, KOH serves as the limiting agent and as a result significant N content was retained in the carbons, which limited the overall porosity.

The level of porosity in the carbons can be interpreted as follows: according to eqn (1), K_2CO_3 interacts with N-C on the surface of the ACDS carbon to form N adducts (KOCN) at *ca.* 600 °C along with generating some porosity from N, and C etching. Sevilla *et al.*⁶³ found that the intermediate KOCN is a reactive form, and upon further heating, the KOCN works as an oxidising agent and can create significant porosity, as per

eqn (2). Thus, the enhancement observed in the mesopore region takes place at 700 °C, which is the temperature at which the decomposition of K_2CO_3 with N-C occurs. This is accompanied by a decrease in yield from 40% at 600 °C (sample DSM2600-1) to 31% at 700 °C (sample DSM2700-1), and a reduction in N content (Table 1). Further increase in activation temperature from 700 to 800 °C largely enhances the proportion of pores of size ~ 3 nm, accompanied by a drop in the carbon yield to 16% for DSM4800-1. As more N atoms are involved in the reaction and removed by the etching reaction with K_2CO_3 , larger pores are generated alongside a small proportion of



Table 2 Yield and/or elemental composition of raw date seeds, air-carbonised date seeds (ACDS) and activated carbons prepared at melamine or urea/ACDS ratio of 2

Sample	Yield [wt%]	C [%]	H [%]	N [%]	O [%]	(O/C) ^a	(N/C) ^a
Raw DS	—	49.4	6.0	1.1	43.5	0.660	0.019
ACDS	50	76.7	3.8	2.0	17.5	0.171	0.022
DSM-2	—	43.7	4.2	47.5	4.6	0.079	0.932
DSU-2	—	38.2	5.8	33.6	22.4	0.440	0.754
DSM2600-2	33	67.7	1.5	18.0	12.8	0.142	0.228
DSM2700-2	28	72.5	0.9	14.9	11.7	0.121	0.176
DSM2800-2	23	77.3	0.7	10.1	11.9	0.115	0.112
DSM4600-2	25	74.7	1.3	7.4	16.6	0.167	0.085
DSM4700-2	18	76.0	0.6	7.0	16.4	0.162	0.079
DSM4800-2	10	89.6	0.2	1.2	9.2	0.077	0.012
DSU4800-2	9	90.5	0.0	0.5	8.9	0.074	0.005

^a Atomic ratio

micropores. However, when the N content is very high and a low amount of KOH is used (DSM2T-2 carbons), the limited amount of K₂CO₃ generated works as a limiting factor, resulting in excess N content remaining in the resulting carbons. Altogether, in this way, the melamine acts as both an N-dopant and a structure-directing agent (porogen) by extending the size distribution of the pores into the mesopore region.

The textural properties of the N-doped activated carbons in Table 3 show that the surface area and pore volume range from moderate to ultra-high depending on the amount of N-dopant added to the activation mix and the severity of activation. For samples prepared at a melamine/ACDS ratio of 1, the surface area ranges from 1029 m² g⁻¹ for DSM2600-1 to 2536 m² g⁻¹ for DSM2800-1, and from 2226 m² g⁻¹ for DSM4600-1 to 3360 m² g⁻¹ for DSM4800-1, while the equivalent sample prepared in the

presence of urea (DSU4800-1) has the highest surface area of 3646 m² g⁻¹. A similar pattern is observed for the pore volume, which varies from 0.43 to 1.18 cm³ g⁻¹ for DSM2T-1 samples, 1.06 to 1.84 cm³ g⁻¹ for DSM4T-1 samples, and reaches 2.05 cm³ g⁻¹ for DSU4800-1. When the amount of melamine was increased, the surface area and pore volume decreased under any given activation conditions. Thus DSM2T-2 carbons have surface areas ranging from 340 m² g⁻¹ for DSM2600-2 to 1161 m² g⁻¹ for DSM2800-2 and pore volume of 0.15 cm³ g⁻¹ and 0.56 cm³ g⁻¹, respectively. Carbons prepared with a high amount of both melamine and KOH (DSM4T-2 carbons) had higher surface area and pore volume of 1322 m² g⁻¹ and 0.59 cm³ g⁻¹ respectively, for DSM4600-2, which increased to 3138 m² g⁻¹ and 1.75 cm³ g⁻¹ for DSM4800-2. Sample DSU4800-2 has the highest surface area (3270 m² g⁻¹) and pore volume (1.79 cm³ g⁻¹) in this group of samples. In general, the samples prepared in the presence of urea (DSU4800-y) have higher surface area than their counterparts (DSM4800-y) prepared with melamine. This is because melamine has a bulk N content of ~67 wt% compared to 46 wt% for urea, meaning a higher N/K ratio for the former. It is noteworthy that the retained N content follows the trend DSM4800-2 > DSU4800-2 > DSM4800-1 > DSU4800-1, which is consistent with their surface area of 3138, 3270, 3360 and 3646 m² g⁻¹. Thus DSU4800-1 has the lowest N/K ratio among all samples in this study and has the highest surface area and pore volume, which may be attributed to an ideal N/K ratio where significant amounts of N are consumed during the activation. Interestingly, the significant increase in the surface area from modest to ultra-high is consistent with the decreasing N content (N/C) in the samples, which, as described above, means more N was involved in the creation of the greater surface area and porosity.^{64,65}

Table 3 Textural properties of N-doped activated carbons prepared at melamine or urea/ACDS ratio of 1 or 2, KOH/ACDS ratio of 2 or 4, and activation temperature of 600, 700, or 800 °C

Sample	Surface area (m ² g ⁻¹)		Pore volume (cm ³ g ⁻¹)		Surface area density ^c (m ² cm ⁻³)	Packing density ^d (g cm ⁻³)	Volumetric surface area ^e (m ² cm ⁻³)
	Total	Micro. ^a (%)	Total	Micro. ^b (%)			
DSM2600-1	1029	891 (87)	0.43	0.35 (82)	2393	0.89	916
DSM2700-1	1859	1441 (78)	0.82	0.58 (71)	2267	0.74	1376
DSM2800-1	2536	1513 (60)	1.18	0.60 (57)	2149	0.65	1648
DSM4600-1	2226	1690 (76)	1.06	0.67 (63)	2100	0.68	1514
DSM4700-1	2672	1712 (64)	1.28	0.68 (53)	2088	0.59	1577
DSM4800-1	3360	510 (15)	1.84	0.15 (8)	1826	0.45	1512
DSU4800-1	3646	578 (16)	2.05	0.19 (9)	1779	0.41	1495
DSM2600-2	340	295 (87)	0.15	0.12 (80)	2267	—	—
DSM2700-2	635	484 (76)	0.30	0.20 (67)	2117	—	—
DSM2800-2	1161	756 (65)	0.56	0.32 (57)	2073	—	—
DSM4600-2	1322	1113 (84)	0.59	0.44 (75)	2241	—	—
DSM4700-2	1989	1383 (70)	0.93	0.57 (61)	2139	—	—
DSM4800-2	3138	400 (13)	1.75	0.14 (8)	1793	0.43	1349
DSU4800-2	3270	366 (11)	1.79	0.11 (6)	1827	0.41	1341

^a The values in parentheses are the proportion of micropore surface area. ^b The values in parentheses are the proportion of micropore volume. ^c Surface area density (SAD) defined as the ratio of surface area to pore volume. ^d To determine the packing density, a known quantity of carbon was compressed in a 1.3 cm die at 370 MPa for 10 minutes. ^e Multiplying the surface area by the packing density yields the volumetric surface area.



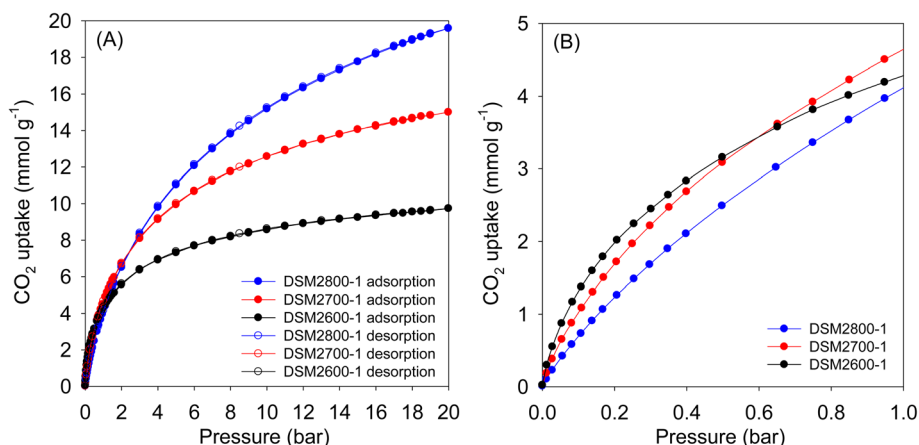


Fig. 5 CO₂ uptake at 25 °C of N-doped activated carbons prepared at melamine/ACDS ratio of 1 and KOH/ACDS ratio of 2 in the pressure range of (A) 0–20 bar, and (B) 0–1 bar.

Table 4 CO₂ uptake at 25 °C and various pressures (0.15, 1, and 20 bar) of N-doped activated carbons prepared at melamine or urea/ACDS ratio of 1 and KOH/ACDS ratio of 2 or 4

Sample	CO ₂ uptake (mmol g ⁻¹)		
	0.15 bar	1 bar	20 bar
DSM2600-1	1.7	4.3	9.8
DSM2700-1	1.4	4.6	15.0
DSM2800-1	1.0	4.1	19.6
DSM4600-1	1.1	4.1	17.9
DSM4700-1	0.9	4.0	21.3
DSM4800-1	0.6	3.2	24.9
DSU4800-1	0.6	3.2	24.7

The surface area density (SAD) of the carbons, defined as the ratio of surface area to pore volume, is shown in Table 3. The SAD shows a clear correlation with the nature of porosity, and

decreases at higher levels of mesoporosity. For melamine/ACDS ratio of 1, the SAD ranges between 2393 and 2149 m² cm⁻³ for DSM2T-1 carbons, and between 2100 m² cm⁻³ and 1826 m² cm⁻³ for DSM4T-1 carbons. For melamine/ACDS ratio of 2, the SAD ranges from 2267 to 2073 m² cm⁻³ for DSM2T-2 carbons and 2241 to 1793 m² cm⁻³ for DSM4T-2 samples. Sample DSU4800-1, which has the highest surface area, had the lowest surface area density of 1779 m² cm⁻³ due to a high level of mesoporosity.

3.4 Gas uptake

3.4.1 CO₂ uptake. Given the broad range of porosity for the present carbons, their potential use as stores for carbon capture and storage (CCS) was investigated at 25 °C and a pressure range of 0–20 bar. Fig. 5 and 6 show the CO₂ uptake isotherms for DSMxT-1 carbons along with sample DSU4800-1, and Table 4 summarises the CO₂ uptake at various pressures. The

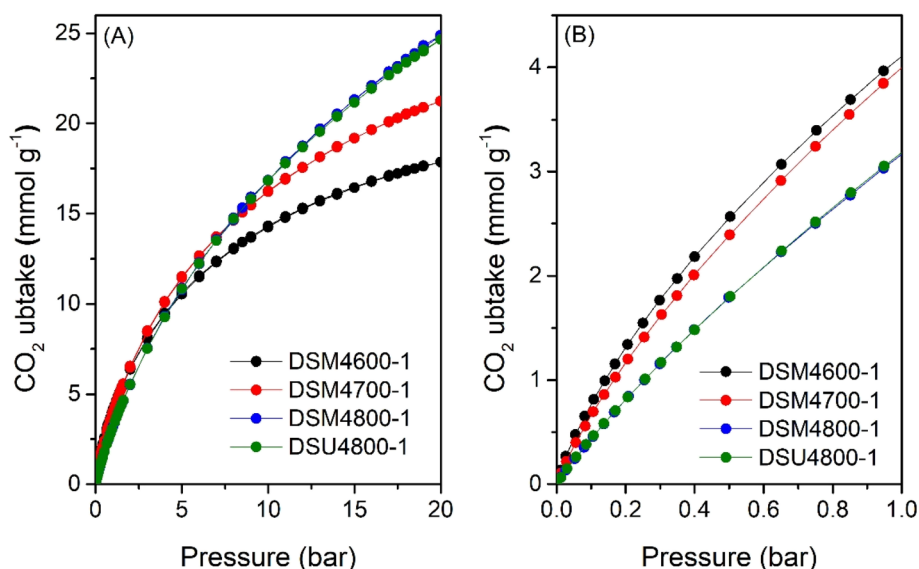


Fig. 6 CO₂ uptake at 25 °C of N-doped activated carbons prepared at melamine or urea/ACDS ratio of 1 and KOH/ACDS ratio of 4 in the pressure range of (A) 0–20 bar, and (B) 0–1 bar.



CO₂ uptake at 1 bar (Fig. 5B) for DSM2T-1 carbons is very attractive and ranges from 4.3 mmol g⁻¹ (DSM2600-1) to 4.6 mmol g⁻¹ (DSM700-1) with a decrease to 4.1 mmol g⁻¹ for DSM2800-1. DSM4T-1 carbons (Fig. 6B), have lower CO₂ uptake (3.2–4.1 mmol g⁻¹) due to low levels of microporosity, which confirms that uptake at low pressure is dependent on pore size rather than the overall surface area.^{35,36,66} At 20 bar, however, the situation is reversed; the storage capacity of DSM4T-1 carbons is higher, up to ca. 25 mmol g⁻¹ for DSM4800-1 and DSU4800-1, due to the greater dependence of uptake on the total surface area. At 0.15 bar, DSM2T-1 carbons show attractive CO₂ uptake of up to 1.7 mmol g⁻¹ for DSM2600-1, which is amongst the highest reported for any porous material (ESI Table S1†).^{28,66–83} The uptake for DSM4T-1 samples is lower and ranges between 0.6 and 1.1 mmol g⁻¹ due to lower levels of microporosity. We do not observe any clear relationship between the N content and CO₂ uptake, meaning that the porosity plays the key role. For the present samples, the N-dopants direct the porosity of the

Table 5 CO₂ uptake at 25 °C and various pressures (0.15, 1 and 20 bar) of N-doped activated carbons prepared at melamine or urea/ACDS ratio of 2 and KOH/ACDS ratio of 2 or 4

Sample	CO ₂ uptake (mmol g ⁻¹)		
	0.15 bar	1 bar	20 bar
DSM2600-2	1.2	2.5	4.5
DSM2700-2	1.3	3.0	6.6
DSM2800-2	1.3	3.4	9.6
DSM4600-2	1.6	4.7	12.2
DSM4700-2	1.2	4.3	16.0
DSM4800-2	0.7	3.3	23.4
DSU4800-2	0.7	3.2	24.4

activated carbons rather than any direct impact on the CO₂ uptake due to the presence of N-moieties.

The CO₂ uptake isotherms of DSM_xT-2 carbons and DSU4800-2 are shown in Fig. 7, 8, and Table 5 summarises the

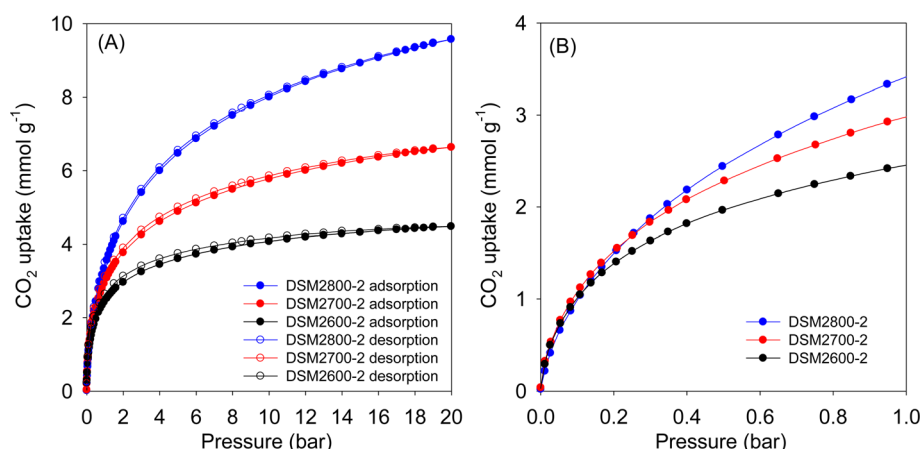


Fig. 7 CO₂ uptake at 25 °C of N-doped activated carbons prepared at melamine/ACDS ratio of 2 and KOH/ACDS ratio of 2 in the pressure range of (A) 0–20 bar, and (B) 0–1 bar.

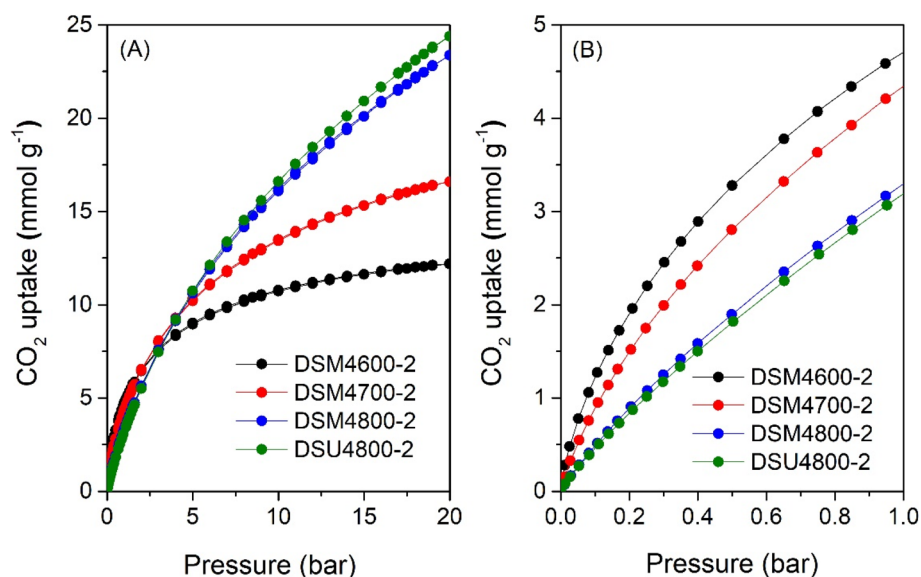


Fig. 8 CO₂ uptake at 25 °C of N-doped activated carbons prepared at melamine or urea/ACDS ratio of 2 and KOH/ACDS ratio of 4 in the pressure range of (A) 0–20 bar, and (B) 0–1 bar.



CO₂ storage capacity at various pressures. As shown in Fig. 7A, the isotherms are fully reversible, which is consistent with physisorption of CO₂ on the carbons. This also rules out any possible trend towards stronger adsorption of CO₂ due to the N-doping. At 1 bar, the uptake is 2.5 and 3.4 mmol g⁻¹ for DSM2600-2 and DSM2800-2, respectively. Carbons derived under harsher activation conditions (DSM4T-2) show uptake of ~3.2 mmol g⁻¹ for DSM4800-2 and DSU4800-2, which rises to 4.7 and 4.3 mmol g⁻¹ for samples obtained at lower activation temperatures of 600 °C and 700 °C, respectively. All the samples approach saturation at 20 bar (Fig. 7A and 8A) with the exception of DSM4800-2 and DSU4800-2. The low uptake for some samples at 20 bar is attributed to the use of high amounts of melamine, which generated lowly porous but highly microporous and N-rich carbons.⁶⁵ On the other hand, the CO₂ uptake at very low pressure of 0.15 bar (Fig. 7B and 8B) is attractive and in a narrow range between 1.2 and 1.3 mmol g⁻¹ for DSM2T-2 carbons. It is interesting to note that the uptake for samples prepared at KOH/ACDS ratio of 2 was similar regardless of activation temperature and N content. The CO₂ uptake for samples activated at a KOH/ACDS ratio of 4 is more varied; 1.6 mmol g⁻¹ for DSM4600-2 and 1.2 mmol g⁻¹ for DSM4700-2, and decreases to 0.7 mmol g⁻¹ for DSM4800-2 and DSU4800-2. The lowering in the uptake for DSM4800-2 and DSU4800-2 is expected due to their wider pores. Notably, DSM4600-2 had the highest CO₂ uptake at 0.15 and 1 bar of 1.6 and 4.7 mmol g⁻¹, respectively. This uptake is amongst the highest ever reported for N-doped carbon materials, which show great promise as post-combustion CO₂ storage materials (ESI Table S1†).^{28,66–83}

3.4.2 Methane storage. Porous materials must possess the appropriate combination of micro- and mesoporosity, large surface area, and high packing density in order to effectively store methane at moderate to high pressure (35–100 bar). The presence of optimal microporosity is important to enhance the packing density, which is key for enhanced volumetric methane uptake, while mesoporosity is essential for improving the working capacity or deliverable methane for pressure swing processes.^{7–10,12,25,36–38} The excess methane uptake was determined at pressures between 0 and 100 bar and 25 °C. The

uptake at 35, 65 and 100 bar is the focus of the discussion below as a measure of practical performance that has been extensively used in earlier research, which enables comparisons to the current state-of-the-art materials.^{7–10,12,25,36–38} The total methane uptake capacity was calculated from the excess data by considering the methane density and the total pore volume of the carbons according to the following equation:

$$\theta_T = \theta_{\text{Exc}} + d_{\text{CH}_4} \times V_T \quad (3)$$

where: θ_T is total methane uptake (g g⁻¹), θ_{Exc} is excess methane uptake (g g⁻¹), d_{CH_4} is methane density (g cm⁻³) at any given pressure at 25 °C, and V_T is total pore volume (cm³ g⁻¹) of the activated carbon. The density of methane (d_{CH_4}) was obtained from the National Institute of Standards and Technology (NIST) website (<https://www.nist.gov/>).

Fig. 9 shows the excess methane uptake isotherms at 25 °C for a selection of samples, and the storage capacity at 35, 65, and 100 bar is summarised in Table 6. Fig. 9A shows the methane uptake of representative DSMxT-1 carbons that have

Table 6 Excess gravimetric methane uptake at 25 °C and pressures of 35, 65 and 100 bar of N-doped activated carbons prepared at KOH/ACDS ratio of 2 or 4 and melamine or urea/ACDS ratio of 1 or 2

Sample	Excess gravimetric methane uptake (mmol g ⁻¹) or (g g ⁻¹) ^a		
	35 bar	65 bar	100 bar
DSM2700-1	8.5 (0.14)	9.5 (0.15)	9.8 (0.16)
DSM2800-1	10.9 (0.17)	12.4 (0.20)	12.7 (0.21)
DSM4600-1	9.5 (0.15)	10.8 (0.17)	11.0 (0.18)
DSM4700-1	11.1 (0.18)	12.8 (0.21)	13.3 (0.22)
DSM4800-1	12.5 (0.20)	14.8 (0.24)	15.5 (0.25)
DSM4800-2	12.2 (0.19)	14.5 (0.23)	15.2 (0.24)
DSU4800-1	13.0 (0.21)	15.5 (0.25)	16.3 (0.26)
DSU4800-2	12.4 (0.20)	14.5 (0.23)	14.8 (0.24)

^a The values in parentheses are the excess gravimetric methane uptake (g g⁻¹).

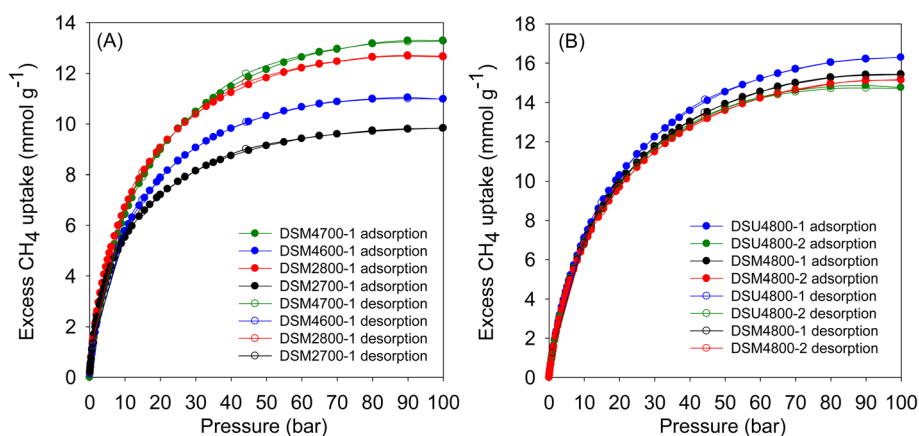


Fig. 9 Excess gravimetric methane uptake at 25 °C of N-doped activated carbons prepared at (A) KOH/ACDS ratio of 2 or 4 and melamine/ACDS ratio of 1, or (B) KOH/ACDS ratio of 4 and melamine or urea/ACDS ratio of 1 or 2.



sufficiently high surface area to be of interest as methane stores. These samples have high methane storage capacity, with greater storage for samples that have higher surface area in the order DSM2700-1 < DSM4600-1 < DSM2800-1 < DSM4700-1. Thus, the excess uptake at 35 bar is in the range of 8.5–10.9 mmol g⁻¹, which increased to 9.5–12.4 and 9.8–12.7 mmol g⁻¹ at 65 and 100 bar, respectively. On the other hand, the excess methane uptake of samples obtained under more severe activation conditions (DSM4800-1, DSM4800-2, DSU4800-1 and DSU4800-2), show higher methane uptake (Fig. 9B and Table 6). The methane storage capacity of these carbons is in very narrow ranges; 12.2 and 13.0 mmol g⁻¹ at 35 bar, 14.5 to 15.5 mmol g⁻¹ at 65 bar, and 14.8 to 16.3 mmol g⁻¹ at 100 bar. The excess methane uptake of 13.0 mmol g⁻¹ at 25 °C and 35 bar for DSU4800-1 is among the highest ever reported for any porous materials (ESI Table S2†).^{7–12,15,36–38,84–87} At 100 bar, the highest excess methane uptake is for samples with the highest surface area (DSU4800-1 and DSM4800-1) at 16.3 and 15.5 mmol g⁻¹, respectively. It may, therefore, be inferred that samples prepared with a low N/K ratio store more methane than those prepared at higher ratio. This is an indication that the strategy of adding N-containing additives, as dopants and structure-directing agents, to the ACDS precursor enhances the methane storage capacity, especially when compared to the uptake of carbons prepared from ACDS carbon only.³⁸

Fig. 10 shows the total gravimetric methane uptake isotherms of samples DSM2700-1, DSM2800-1, DSM4600-1, DSM4700-1, DSM4800-1, and DSM4800-2, as well as those of DSU4800-1 and DSU4800-2. Table 7 provides a summary of the total gravimetric methane uptake at 35, 65, and 100 bar. The total methane storage for carbons produced under milder conditions is 9.7, 12.6, 11.1 and 13.0 mmol g⁻¹ (at 35 bar), 11.9, 15.8, 13.9 and 16.6 mmol g⁻¹ (at 65 bar), and 13.7, 18.3, 16.0 and 19.4 mmol g⁻¹ (at 100 bar), respectively, for DSM2700-1,

Table 7 Total gravimetric methane uptake at 25 °C and pressures of 35, 65 and 100 bar of N-doped activated carbons prepared at KOH/ACDS ratio of 2 or 4 and melamine or urea/ACDS ratio of 1 or 2

Sample	Total gravimetric methane uptake (mmol g ⁻¹) or (g g ⁻¹) ^a		
	35 bar	65 bar	100 bar
DSM2700-1	9.7 (0.16)	11.9 (0.19)	13.7 (0.22)
DSM2800-1	12.6 (0.20)	15.8 (0.25)	18.3 (0.29)
DSM4600-1	11.1 (0.18)	13.9 (0.22)	16.0 (0.26)
DSM4700-1	13.0 (0.21)	16.6 (0.27)	19.4 (0.31)
DSM4800-1	15.3 (0.25)	20.2 (0.32)	24.2 (0.39)
DSM4800-2	14.8 (0.24)	19.6 (0.31)	23.5 (0.38)
DSU4800-1	16.1 (0.26)	21.5 (0.35)	26.0 (0.42)
DSU4800-2	15.1 (0.24)	19.7 (0.32)	23.4 (0.37)

^a The values in parentheses are the total gravimetric methane uptake donated as (g g⁻¹).

DSM2800-1, DSM4600-1 and DSM4700-1. At 35 bar, the total methane uptake is 15.3 and 14.8 mmol g⁻¹ for DSM4800-1 and DSM4800-2, respectively, while for DSU4800-1 and DSU4800-2, the total uptake is 16.1 and 15.1 mmol g⁻¹, respectively. At 65 bar, the total methane uptake rises to 20.2 and 19.6 mmol g⁻¹ for DSM4800-1 and DSM4800-2, respectively, and increases further to 24.2 and 23.5 mmol g⁻¹ at 100 bar. While the total uptake is 21.5 and 19.7 mmol g⁻¹ at 65 bar, and 26.0 and 23.4 mmol g⁻¹ at 100 bar for DSU4800-1 and DSU4800-2, respectively. It is noteworthy that at 100 bar, the uptake of some carbons (*i.e.*, 0.39 and 0.38 g g⁻¹ for DSM4800-1 and DSM4800-2, and 0.42 and 0.37 g g⁻¹ for DSU4800-1 and DSU4800-2, respectively) is close to the DOE target of 0.5 g g⁻¹. The N-doping process, therefore, generates carbons with attractive gravimetric methane uptake that either matches or

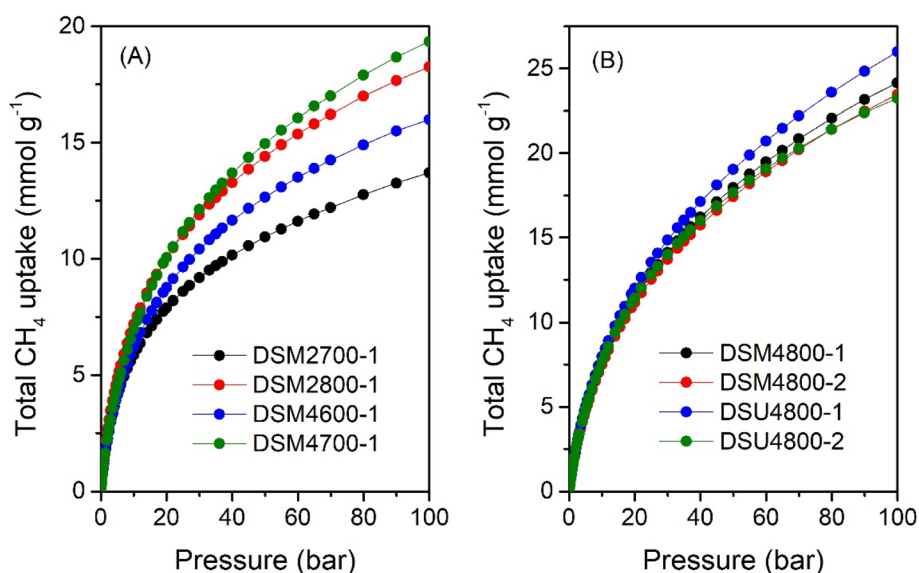


Fig. 10 Total gravimetric methane uptake at 25 °C of N-doped activated carbons prepared at (A) KOH/ACDS ratio of 2 or 4 and melamine/ACDS ratio of 1, or (B) KOH/ACDS ratio of 4 and melamine or urea/ACDS ratio of 1 or 2.



exceeds those previously reported to date for carbon or MOF materials (ESI Table S2†).^{7–14,37,38,84,87–92} Furthermore, the gravimetric uptake of all the present carbons is higher than that of carbons prepared from ACDS only.³⁸

The volumetric uptake (*i.e.*, cm^3 (STP) cm^{-3}), which is the amount of methane that can be held by an adsorbent per unit of tank volume, is an important parameter for methane storage performance from a practical perspective. The volumetric uptake is determined by the packing density of the adsorbent and the gravimetric uptake. If an adsorbent has a higher packing density, it could potentially allow more of it to be filled in a given tank, which in turn increases the volumetric capacity. The United States Department of Energy (U.S. DOE) has established a target for volumetric methane uptake at 25 °C and moderate pressure (*i.e.*, 35–100 bar) of 263 cm^3 (STP) cm^{-3} . The total volumetric methane uptake isotherms are shown in Fig. 11, and Table 8 provides a summary of storage capacity at 35, 65 and 100 bar. The volumetric uptake isotherms do not show any saturation at 100 bar, which suggests that the carbons could store greater amounts of methane above 100 bar. This contrasts with most benchmark MOFs that reach saturation at pressures of *ca.* 80 bar.^{14,93} For carbons obtained at milder activation conditions (DSM2700-1, DSM2800-1, DSM4600-1 and DSM4700-1), at 35 bar, the total volumetric methane uptake (cm^3 (STP) cm^{-3}) varies from 161 to 184, while it is between 154 and 138 for those obtained under harsher conditions (DSM4800-*y* and DSU4800-*y*). The highest volumetric methane uptake capacity at 35 bar, observed for DSM2800-1 at 184 cm^3 (STP) cm^{-3} , is amongst the best reported for porous materials.^{8,16,37,94,95} At 65 bar, the uptake is between 198 and 230 cm^3 (STP) cm^{-3} for DSM2700-1, DSM2800-1, DSM4600-1 and DSM4700-1, and in the range of 181 to 203 cm^3 (STP) cm^{-3} for DSM4800-*y* and DSU4800-*y*. At 100 bar, DSM2700-1, DSM2800-1, DSM4600-1, and DSM4700-1

Table 8 Total volumetric methane uptake and working capacity of N-doped activated carbons prepared at KOH/ACC ratio of 2 or 4 and melamine or urea/ACDS ratio of 1 or 2^a

Sample	Total volumetric methane uptake (cm^3 (STP) cm^{-3})		
	35 bar	65 bar	100 bar
DSM2700-1	161 (92)	198 (129)	227 (158)
DSM2800-1	184 (113)	230 (159)	266 (195)
DSM4600-1	169 (105)	212 (148)	244 (180)
DSM4700-1	171 (111)	219 (159)	256 (196)
DSM4800-1	154 (105)	203 (154)	244 (195)
DSM4800-2	143 (97)	189 (143)	226 (180)
DSU4800-1	148 (102)	197 (151)	239 (193)
DSU4800-2	138 (93)	181 (136)	214 (169)

^a The values in parentheses are volumetric working capacity (*i.e.*, deliverable methane), which is calculated by subtracting the storage pressure at 5 bar.

carbons have total methane uptake of 227, 266, 244, and 256 cm^3 (STP) cm^{-3} , respectively, and DSM4800-1, DSM4800-2, and DSU4800-1 and DSU4800-2 have total methane uptake of 244, 226, 239, and 214 cm^3 (STP) cm^{-3} . The best volumetric uptake at 100 bar is for sample DSM2800-1 at 266 cm^3 (STP) cm^{-3} , which meets the DOE target and outperforms or is comparable to the best activated carbons or MOFs (ESI Table S2†).^{8,12,37,84} On the other hand, it is interesting to note that, at 25 °C and 100 bar, the uptake of DSU4800-1 is only ~9% lower than the DOE target values (0.5 g g^{-1} and 263 cm^3 (STP) cm^{-3}) for gravimetric and volumetric uptake, *i.e.*, 0.42 g g^{-1} and 239 cm^3 (STP) cm^{-3} , respectively. The attractive performance of the DSU4800-1 samples can be attributed to a well-balanced combination of porosity and packing density as a result of an optimal N:K ratio during synthesis. Overall, the volumetric

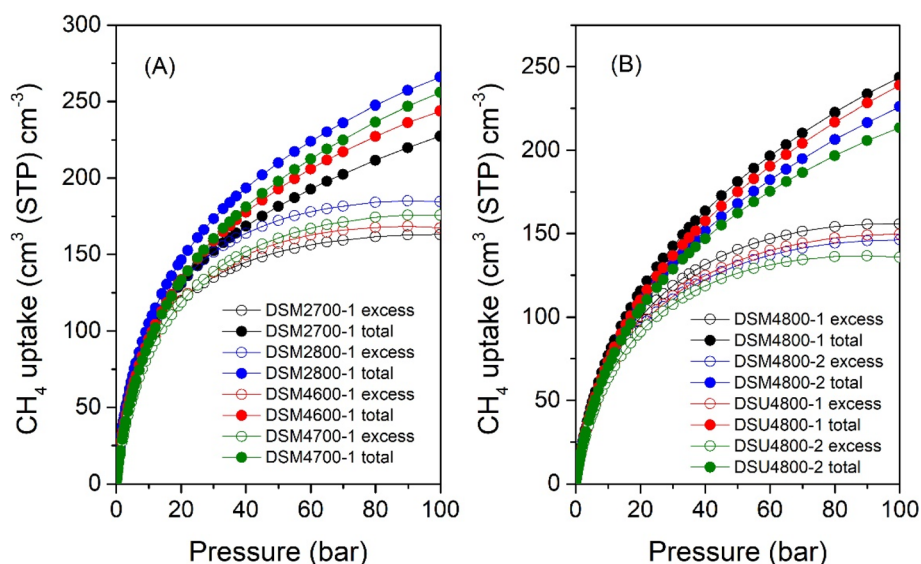


Fig. 11 Excess and total volumetric methane uptake isotherms at 25 °C of N-doped activated carbons prepared at (A) KOH/ACDS ratio of 2 or 4 and melamine/ACDS ratio of 1, or (B) KOH/ACDS ratio of 4 and melamine or urea/ACDS ratio of 1 or 2.



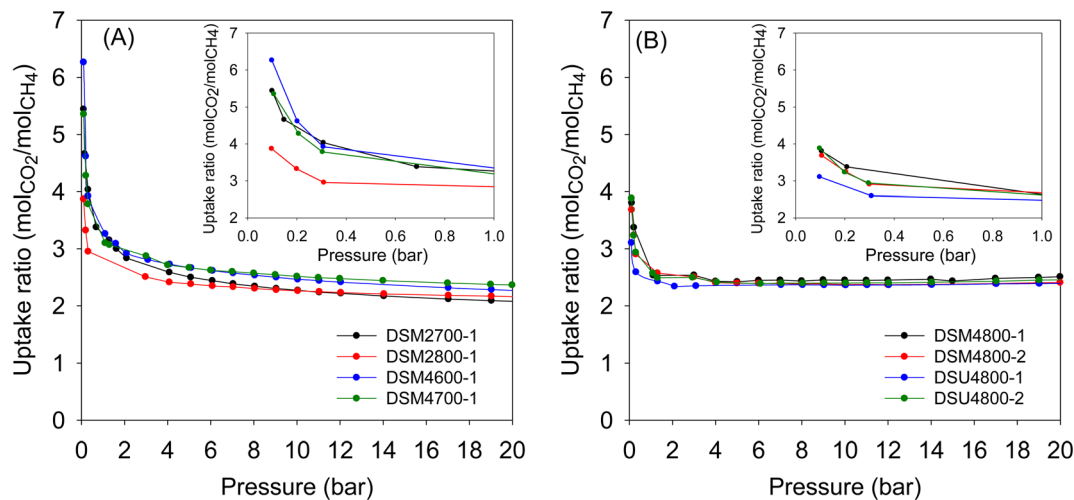


Fig. 12 Uptake ratio of CO₂/CH₄ as a function of pressure for a 50/50 mix of the gases for N-doped activated carbons prepared at (A) KOH/ACDS ratio of 2 or 4 and melamine/ACDS ratio of 1, or (B) KOH/ACDS ratio of 4 and melamine or urea/ACDS ratio of 1 or 2.

uptake obtained by the activated carbons in this study is attractive in the context of previous reports on carbons or MOFs (ESI Table S2†).^{7–11,15,16,37,38,85–87,94–100}

The working capacity, *i.e.*, the efficacy of materials to deliver the stored gas between two finite pressures, is more important than the total storage capacity. The efficiency of this process is largely dependent on the type of porosity. Highly microporous materials absorb a large amount of methane at lower pressures (*e.g.*, 5 bar), thereby reducing their working capacity. Thus, to achieve high working capacity, the mix of microporosity and mesoporosity is important; the former is critical for high packing density, while the latter plays an important role in minimising the uptake at lower pressure, leading to an increase in the working capacity of the materials. Deliverable methane, or volumetric working capacity, calculated by subtracting the storage at release pressure of 5 bar from the uptake at 35, 65 or 100 bar, is given in Table 8. The volumetric working capacity, for a 100 to 5 bar pressure swing, reaches 196, 195, 195, and 193 cm³ (STP) cm⁻³ for samples DSM4700-1, DSM2800-1, DSM4800-1 and DSU4800-1, respectively. It's interesting that the volumetric capacities of DSM4800-1 and DSU4800-1 are almost the same as for DSM2800-1, even though the later has the highest total volumetric uptake of 266 cm³ (STP) cm⁻³. This is ascribed to the presence of greater mesoporosity in DSM4800-1 and DSU4800-1, which lowers the uptake at 5 bar.⁹³ It is noteworthy that the present carbons have volumetric working capacities approaching the values of the best MOFs (*i.e.*, monolithic mono-HKUST1 and mono-UiO-66_D), which are considered to be 50% better than any powder MOF (ESI Table S3†).^{12,98}

Although the main focus of this work was the storage of CO₂ and methane, we also explored the performance in the N-doped activated carbons in the separation of CO₂/CH₄ mixtures. Such separation is important especially for the purification of natural gas or biomethane, where CO₂ and other impurities are removed, prior to use as a fuel. The porosity of the present carbons means that they adsorb more CO₂ compared to

methane (ESI Fig. S9†). The relative uptake of CO₂ and methane depends on the porosity of the carbons. By assuming a 50/50 mix of CO₂ and methane, we are able to estimate the CO₂/CH₄ selectivity based on the uptake ratio. Fig. 12 shows the CO₂/CH₄ uptake ratio as a function of pressure in the range 0–20 bar. As expected, the CO₂/CH₄ uptake ratio is highest at pressures close to zero and then reduces at higher pressure. The more microporous carbons (Table 3) generally achieve higher CO₂/CH₄ uptake ratio. These carbons can therefore be effective for the enrichment of methane from CO₂/CH₄ gas streams.

4. Conclusions

N-doped carbons with well controlled porosity were prepared *via* the addition of N-containing additives (melamine or urea) to an activation mixture containing KOH as an activating agent and carbonaceous matter derived from biomass, *i.e.*, air-carbonised date seed, *Phoenix dactylifera*, ACDS. Depending on the preparation conditions, and by varying the amount of melamine or urea, the amount of KOH, and the activation temperature, carbons with a wide range of porosity were generated. The activated carbons feature a large N content, which can reach up to 18 wt%, with surface area of between 340 to 3646 m² g⁻¹ and pore volume in the range of 0.15 to 2.1 cm³ g⁻¹. It was observed that the N added to the activation mix serves as both an N-dopant and a structure-directing agent (porogen) by extending the size of pores generated into the mesopore region. This study provides evidence that N-containing additives can play a key role determining the structural properties of the activated carbons, and their use in this study acted to increase the susceptibility of the biomass-derived ACDS carbon to activation. It appears that K⁺ from the KOH reacts with surface N species readily, leading to a significant increase in the surface area, which is accompanied with a lowering of the N content of the carbons. Furthermore, the presence of N provides activation pathways to carbons with a unique combination of textural properties (*i.e.*, surface area,



mix of microporosity and mesoporosity and packing density) that can be tuned towards suitability for enhanced uptake of CO₂ and/or methane. Depending on the preparation conditions and resulting mix of micro/mesoporosity, the carbons show excellent low pressure CO₂ capture at 25 °C of 1.7 mmol g⁻¹ at 0.15 bar and 4.7 mmol g⁻¹ at 1 bar. At 25 °C and 100 bar the methane uptake of the carbons reaches 0.42 g g⁻¹ and 266 cm³ (STP) cm⁻³, respectively, and working capacity of 196 cm³ (STP) cm⁻³ for a 100 to 5 bar pressure swing. The impressive performance is attributed to the balanced combination of porosity and packing density as a result of an optimal N : K ratio during the activation process.

Data availability

Data that support the findings of this study are available within the paper (and its ESI Files†) and from the corresponding author on reasonable request.

Author contributions

N. A.: methodology, formal analysis, investigation, writing. R. M.: conceptualisation, writing – review & editing, supervision, funding acquisition, resources.

Conflicts of interest

There are no conflicts to declare.

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References

- J. Rogelj, D. Shindell, K. Jiang, S. Fifita, P. Forster, V. Ginzburg, C. Handa, H. Khesghi, S. Kobayashi and E. Kriegler, Global warming of 1.5 °C. An IPCC special report on the impacts of global warming of 1.5 °C above pre-industrial levels and related global greenhouse gas emission pathways, in *The Context of Strengthening the Global Response to the Threat of Climate Change, Sustainable Development, and Efforts to Eradicate Poverty*, 2018.
- International Energy Agency (IEA), *Global CO₂ Emissions Rebounded to their Highest Level in History in 2021*, <https://www.iea.org/news/global-co2-emissions-rebounded-to-their-highest-level-in-history-in-2021>, accessed 8 March 2022.
- C. F. Schleussner, J. Rogelj, M. Schaeffer, T. Lissner, R. Licker, E. M. Fischer, R. Knutti, A. Levermann, K. Frieler and W. Hare, *Nat. Clim. Change*, 2016, **6**, 827–835.
- L. Shi, K.-Y. Liao, Y.-H. Dong, Y.-A. Wang, Y. Zhou, X.-G. Yi, M.-S. Sun, W. Hui and D.-J. Tao, *Sustainable Mater. Technol.*, 2024, **40**, e00880.
- S. Choi, J. H. Drese and C. W. Jones, *ChemSusChem*, 2009, **2**, 796–854.
- G. P. Peters, R. M. Andrew, J. G. Canadell, S. Fuss, R. B. Jackson, J. I. Korsbakken, C. Le Quere and N. Nakicenovic, *Nat. Clim. Change*, 2017, **7**, 118–122.
- T. A. Makal, J. R. Li, W. Lu and H. C. Zhou, *Chem. Soc. Rev.*, 2012, **41**, 7761–7779.
- J. A. Mason, M. Veenstra and J. R. Long, *Chem. Sci.*, 2014, **5**, 32–51.
- K. V. Kumar, K. Preuss, M. M. Titirici and F. Rodriguez-Reinoso, *Chem. Rev.*, 2017, **117**, 1796–1825.
- B. Li, H. M. Wen, W. Zhou, J. Q. Xu and B. L. Chen, *Chem*, 2016, **1**, 557–580.
- Y. He, W. Zhou, G. Qian and B. Chen, *Chem. Soc. Rev.*, 2014, **43**, 5657–5678.
- T. Tian, Z. Zeng, D. Vulpe, M. E. Casco, G. Dovitini, P. A. Midgley, J. Silvestre-Albero, J. C. Tan, P. Z. Moghadam and D. Fairen-Jimenez, *Nat. Mater.*, 2018, **17**, 174–179.
- Y. C. Lin, C. L. Kong, Q. J. Zhang and L. Chen, *Adv. Energy Mater.*, 2017, **7**, 1601296.
- Y. Peng, V. Krungleviciute, I. Eryazici, J. T. Hupp, O. K. Farha and T. Yildirim, *J. Am. Chem. Soc.*, 2013, **135**, 11887–11894.
- S. Dutta, A. Bhaumik and K. C. W. Wu, *Energy Environ. Sci.*, 2014, **7**, 3574–3592.
- M. E. Casco, M. Martínez-Escandell, E. Gadea-Ramos, K. Kaneko, J. Silvestre-Albero and F. Rodríguez-Reinoso, *Chem. Mater.*, 2015, **27**, 959–964.
- M. Namvar-Asl, M. Soltanieh and A. Rashidi, *Energy Convers. Manage.*, 2008, **49**, 2478–2482.
- M. Feroldi, A. C. Neves, C. E. Borba and H. J. Alves, *J. Clean Prod.*, 2018, **172**, 921–926.
- M. Sevilla and R. Mokaya, *Energy Environ. Sci.*, 2014, **7**, 1250–1280.
- J. C. Wang and S. Kaskel, *J. Mater. Chem.*, 2012, **22**, 23710–23725.
- L. Wei and G. Yushin, *Nano Energy*, 2012, **1**, 552–565.
- M. M. Titirici, R. J. White, C. Falco and M. Sevilla, *Energy Environ. Sci.*, 2012, **5**, 6796–6822.
- Y. D. Xia, Z. X. Yang and Y. Q. Zhu, *J. Mater. Chem. A*, 2013, **1**, 9365–9381.
- L. S. Blankenship, N. Balahmar and R. Mokaya, *Nat. Commun.*, 2017, **8**, 1545.
- L. S. Blankenship and R. Mokaya, *Mater. Adv.*, 2022, **3**, 1905–1930.
- M. Sevilla, A. B. Fuertes and R. Mokaya, *Int. J. Hydrogen Energy*, 2011, **36**, 15658–15663.
- E. S. Hanley, J. P. Deane and B. P. O. Gallachoir, *Renew. Sustainable Energy Rev.*, 2018, **82**, 3027–3045.
- N. P. Wickramaratne and M. Jaroniec, *J. Mater. Chem. A*, 2013, **1**, 112–116.
- B. Adeniran, E. Masika and R. Mokaya, *J. Mater. Chem. A*, 2014, **2**, 14696–14710.



- 30 N. Balahmar, A. C. Mitchell and R. Mokaya, *Adv. Energy Mater.*, 2015, **5**, 1500867.
- 31 B. Adeniran and R. Mokaya, *J. Mater. Chem. A*, 2015, **3**, 5148–5161.
- 32 G. Singh, K. S. Lakhi, S. Sil, S. V. Bhosale, I. Kim, K. Albahily and A. Vinu, *Carbon*, 2019, **148**, 164–186.
- 33 B. Adeniran and R. Mokaya, *Nano Energy*, 2015, **16**, 173–185.
- 34 W. Sangchoom, D. A. Walsh and R. Mokaya, *J. Mater. Chem. A*, 2018, **6**, 18701–18711.
- 35 E. Haffner-Staton, N. Balahmar and R. Mokaya, *J. Mater. Chem. A*, 2016, **4**, 13324–13335.
- 36 I. Alali and R. Mokaya, *Energy Environ. Sci.*, 2022, **15**, 4710–4724.
- 37 A. Altwala and R. Mokaya, *J. Mater. Chem. A*, 2022, **10**, 13744–13757.
- 38 A. Altwala and R. Mokaya, *Energy Environ. Sci.*, 2020, **13**, 2967–2978.
- 39 O. F. Cruz Jr, J. Silvestre-Albero, M. E. Casco, D. Hotza and C. R. Rambo, *Mater. Chem. Phys.*, 2018, **216**, 42–46.
- 40 G. K. Parshetti, S. Chowdhury and R. Balasubramanian, *Fuel*, 2015, **148**, 246–254.
- 41 J. Saleem, U. Bin Shahid, M. Hijab, H. Mackey and G. McKay, *Biomass Convers. Biorefin.*, 2019, **9**, 775–802.
- 42 H. M. Wei, J. Chen, N. Fu, H. J. Chen, H. L. Lin and S. Han, *Electrochim. Acta*, 2018, **266**, 161–169.
- 43 E. A. Hirst, A. Taylor and R. Mokaya, *J. Mater. Chem. A*, 2018, **6**, 12393–12403.
- 44 L. Shao, Y. Sang, N. Liu, J. Liu, P. Zhan, J. Huang and J. Chen, *ACS Omega*, 2020, **5**, 17450–17462.
- 45 N. Balahmar, A. S. Al-Jumialy and R. Mokaya, *J. Mater. Chem. A*, 2017, **5**, 12330–12339.
- 46 N. Balahmar and R. Mokaya, *J. Mater. Chem. A*, 2019, **7**, 17466–17479.
- 47 E. Masika and R. Mokaya, *Energy Environ. Sci.*, 2014, **7**, 427–434.
- 48 M. Sevilla, W. Sangchoom, N. Balahmar, A. B. Fuertes and R. Mokaya, *ACS Sustainable Chem. Eng.*, 2016, **4**, 4710–4716.
- 49 M. E. Casco, M. Martinez-Escandell, J. Silvestre-Albero and F. Rodriguez-Reinoso, *Carbon*, 2014, **67**, 230–235.
- 50 M. Sevilla, R. Mokaya and A. B. Fuertes, *Energy Environ. Sci.*, 2011, **4**, 2930–2936.
- 51 M. H. Kim, S. Yun, H. S. Park, J. T. Han, K. B. Kim and K. C. Roh, *J. Mater. Chem. A*, 2015, **3**, 2564–2567.
- 52 M. Sevilla, N. Alam and R. Mokaya, *J. Phys. Chem. C*, 2010, **114**, 11314–11319.
- 53 Y. Zhu, S. Murali, M. D. Stoller, K. J. Ganesh, W. Cai, P. J. Ferreira, A. Pirkle, R. M. Wallace, K. A. Cychosz, M. Thommes, D. Su, E. A. Stach and R. S. Ruoff, *Science*, 2011, **332**, 1537–1541.
- 54 M. Sevilla, G. A. Ferrero, N. Diez and A. B. Fuertes, *Carbon*, 2018, **131**, 193–200.
- 55 M. Sevilla, A. S. M. Al-Jumialy, A. B. Fuertes and R. Mokaya, *ACS Appl. Mater. Interfaces*, 2018, **10**, 1623–1633.
- 56 A. B. Fuertes and M. Sevilla, *Carbon*, 2015, **94**, 41–52.
- 57 M. Cox and R. Mokaya, *Sustainable Energy Fuels*, 2017, **1**, 1414–1424.
- 58 A. E. Ogungbenro, D. V. Quang, K. A. Al-Ali, L. F. Vega and M. R. M. Abu-Zahra, *J. Environ. Chem. Eng.*, 2018, **6**, 4245–4252.
- 59 A. M. Aljumialy and R. Mokaya, *Mater. Adv.*, 2020, **1**, 3267–3280.
- 60 H. M. Coromina, D. A. Walsh and R. Mokaya, *J. Mater. Chem. A*, 2016, **4**, 280–289.
- 61 X. D. Zhu, Y. C. Liu, F. Qian, C. Zhou, S. C. Zhang and J. M. Chen, *ACS Sustainable Chem. Eng.*, 2015, **3**, 833–840.
- 62 T. S. Blankenship and R. Mokaya, *Energy Environ. Sci.*, 2017, **10**, 2552–2562.
- 63 A. B. Fuertes, G. A. Ferrero and M. Sevilla, *J. Mater. Chem. A*, 2014, **2**, 14439–14448.
- 64 J. V. Guerrero, J. N. Burrow, J. E. Eichler, M. Z. Rahman, M. V. Namireddy, K. A. Friedman, S. S. Coffman, D. C. Calabro and C. B. Mullins, *Energy Fuel*, 2020, **34**, 6101–6112.
- 65 J. E. Eichler, J. N. Burrow, Y. Wang, D. C. Calabro and C. B. Mullins, *Carbon*, 2022, **186**, 711–723.
- 66 M. Sevilla and A. B. Fuertes, *Energy Environ. Sci.*, 2011, **4**, 1765–1771.
- 67 M. Sevilla and A. B. Fuertes, *J. Colloid Interface Sci.*, 2012, **366**, 147–154.
- 68 G. Srinivas, V. Krungleviciute, Z.-X. Guo and T. Yildirim, *Energy Environ. Sci.*, 2014, **7**, 335–342.
- 69 J. Silvestre-Albero, A. Wahby, A. Sepúlveda-Escribano, M. Martínez-Escandell, K. Kaneko and F. Rodríguez-Reinoso, *Chem. Commun.*, 2011, **47**, 6840–6842.
- 70 N. P. Wickramaratne and M. Jaroniec, *ACS Appl. Mater. Interfaces*, 2013, **5**, 1849–1855.
- 71 G. P. Hao, W. C. Li, D. Qian, G. H. Wang, W. P. Zhang, T. Zhang, A. Q. Wang, F. Schuth, H. J. Bongard and A. H. Lu, *J. Am. Chem. Soc.*, 2011, **133**, 11378–11388.
- 72 J. Wang, A. Heerwig, M. R. Lohe, M. Oschatz, L. Borchardt and S. Kaskel, *J. Mater. Chem.*, 2012, **22**, 13911–13913.
- 73 X. Q. Fan, L. X. Zhang, G. B. Zhang, Z. Shu and J. L. Shi, *Carbon*, 2013, **61**, 423–430.
- 74 M. Sevilla, P. Valle-Vigón and A. B. Fuertes, *Adv. Funct. Mater.*, 2011, **21**, 2781–2787.
- 75 W. Xing, C. Liu, Z. Zhou, L. Zhang, J. Zhou, S. Zhuo, Z. Yan, H. Gao, G. Wang and S. Z. Qiao, *Energy Environ. Sci.*, 2012, **5**, 7323–7327.
- 76 Y. D. Xia, R. Mokaya, G. S. Walker and Y. Q. Zhu, *Adv. Energy Mater.*, 2011, **1**, 678–683.
- 77 Y. Zhao, L. Zhao, K. X. Yao, Y. Yang, Q. Zhang and Y. Han, *J. Mater. Chem.*, 2012, **22**, 19726–19731.
- 78 Z. Zhang, J. Zhou, W. Xing, Q. Xue, Z. Yan, S. Zhuo and S. Z. Qiao, *Phys. Chem. Chem. Phys.*, 2013, **15**, 2523–2529.
- 79 M. Nandi, K. Okada, A. Dutta, A. Bhaumik, J. Maruyama, D. Derks and H. Uyama, *Chem. Commun.*, 2012, **48**, 10283–10285.
- 80 M. Saleh, J. N. Tiwari, K. C. Kemp, M. Yousuf and K. S. Kim, *Environ. Sci. Technol.*, 2013, **47**, 5467–5473.
- 81 I. Alali and R. Mokaya, *J. Mater. Chem. A*, 2023, **11**, 6952–6965.
- 82 M. Sevilla, C. Falco, M.-M. Titirici and A. B. Fuertes, *RSC Adv.*, 2012, **2**, 12792–12797.



- 83 A. Altwala and R. Mokaya, *Energy Adv.*, 2022, **1**, 216–224.
- 84 Z. Chen, P. Li, R. Anderson, X. Wang, X. Zhang, L. Robison, L. R. Redfern, S. Moribe, T. Islamoglu, D. A. Gomez-Gualdrón, T. Yildirim, J. F. Stoddart and O. K. Farha, *Science*, 2020, **368**, 297–303.
- 85 P. Pfeifer, L. Aston, M. Banks, S. Barker, J. Burrell, S. Carter, J. Coleman, S. Crockett, C. Faulhaber, J. Flavin, M. Gordon, L. Hardcastle, Z. Kallenborn, M. Kemiki, C. Lapilli, J. Pobst, R. Schott, P. Shah, S. Spellerberg, G. Suppes, D. Taylor, A. Tekeci, C. Wexler, M. Wood, P. Buckley, T. Breier, J. Downing, S. Eastman, P. Freeze, S. Graham, S. Grinter, A. Howard, J. Martinez, D. Radke, T. Vassalli and J. Ilavsky, *Chaos*, 2007, **17**, 041108.
- 86 J. Romanos, S. Sweany, T. Rash, L. Firlej, B. Kuchta, J. C. Idrobo and P. Pfeifer, *Adsorpt. Sci. Technol.*, 2014, **32**, 681–691.
- 87 D. A. Gómez-Gualdrón, C. E. Wilmer, O. K. Farha, J. T. Hupp and R. Q. Snurr, *J. Phys. Chem. C*, 2014, **118**, 6941–6951.
- 88 H. Nishihara and T. Kyotani, *Chem. Commun.*, 2018, **54**, 5648–5673.
- 89 D. Lozano-Castello, D. Cazorla-Amoros and A. Linares-Solano, *Energy Fuel*, 2002, **16**, 1321–1328.
- 90 T. A. Rash, A. Gillespie, B. P. Holbrook, L. H. Hiltzik, J. Romanos, Y. C. Soo, S. Sweany and P. Pfeifer, *Fuel*, 2017, **200**, 371–379.
- 91 J. Jiang, H. Furukawa, Y. B. Zhang and O. M. Yaghi, *J. Am. Chem. Soc.*, 2016, **138**, 10244–10251.
- 92 H. Wu, W. Zhou and T. Yildirim, *J. Am. Chem. Soc.*, 2009, **131**, 4995–5000.
- 93 S. Bracco, D. Piga, I. Bassanetti, J. Perego, A. Comotti and P. Sozzani, *J. Mater. Chem. A*, 2017, **5**, 10328–10337.
- 94 S. Choi, M. A. Alkhabbaz, Y. G. Wang, R. M. Othman and M. Choi, *Carbon*, 2019, **141**, 143–153.
- 95 M. E. Casco, M. Martinez-Escandell, K. Kaneko, J. Silvestre-Albero and F. Rodriguez-Reinoso, *Carbon*, 2015, **93**, 11–21.
- 96 B. M. Connolly, D. G. Madden, A. E. H. Wheatley and D. Fairen-Jimenez, *J. Am. Chem. Soc.*, 2020, **142**, 8541–8549.
- 97 V. Rozyyev, D. Thirion, R. Ullah, J. Lee, M. Jung, H. Oh, M. Atilhan and C. T. Yavuz, *Nat. Energy*, 2019, **4**, 604–611.
- 98 B. M. Connolly, M. Aragonés-Anglada, J. Gandara-Loe, N. A. Danaf, D. C. Lamb, J. P. Mehta, D. Vulpe, S. Wuttke, J. Silvestre-Albero, P. Z. Moghadam, A. E. H. Wheatley and D. Fairen-Jimenez, *Nat. Commun.*, 2019, **10**, 2345.
- 99 N. Albeladi, L. Scott Blankenship and R. Mokaya, *Energy Environ. Sci.*, 2024, **17**, 3060–3076.
- 100 C. M. Simon, J. Kim, D. A. Gomez-Gualdrón, J. S. Camp, Y. G. Chung, R. L. Martin, R. Mercado, M. W. Deem, D. Gunter, M. Haranczyk, D. S. Sholl, R. Q. Snurr and B. Smit, *Energy Environ. Sci.*, 2015, **8**, 1190–1199.

