



## Mixed donor, phenanthroline photoactive MOFs with favourable CO<sub>2</sub> selectivity†

 Cite this: *Chem. Commun.*, 2020, 56, 13377

 Received 23rd August 2020,  
Accepted 2nd September 2020

DOI: 10.1039/d0cc05715k

rsc.li/chemcomm

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**Mixed donor phenanthroline-carboxylate linkers were combined with Mn<sup>II</sup> or Zn<sup>II</sup> to form photoactive MOFs with large pore apertures. The MOFs display high CO<sub>2</sub> adsorption capacities, which consequently causes structural framework flexibility, and align with favorable metrics for selective CO<sub>2</sub> capture. The photophysical properties of the MOFs were investigated, with the Mn<sup>II</sup> MOF giving rise to short triplet LMCT lifetimes.**

Decades of global research into metal-organic framework (MOF) chemistry has garnered a sophisticated understanding of strategies to create framework materials with desired properties. This involves designing frameworks with premediated attributes such as open metal sites (OMS),<sup>1</sup> inherent structural flexibility,<sup>2</sup> tailored pore shapes,<sup>3,4</sup> magnetic behaviour<sup>5</sup> and photoluminescent properties.<sup>6,7</sup> Luminescent MOFs (LMOFs) are rapidly gaining interest due to their promise in a broad range of applications including chemical sensing,<sup>7,8</sup> artificial photosynthetic catalysis<sup>9,10</sup> and optoelectronics.<sup>11</sup> The diverse composition of MOFs means that photoluminescence can arise from a variety of processes.<sup>12</sup> Photon absorption and emission can occur within the organic ligand itself in ligand-centred

emission, or can occur in separate locations in the framework. This leads to metal-metal charge transfer (MMCT), metal-ligand charge transfer (MLCT), ligand-metal charge transfer (LMCT) and ligand-ligand charge transfer (LLCT).<sup>6</sup> Furthermore, the ordered crystalline structure and porous nature of MOFs causes analyte-MOF distances to be constrained, allowing for guest-centred emission which drastically improves sensing performances.<sup>6,13</sup>

Owing to its planarity, aromatic nature and chelating capability, organic linkers incorporating a 1,10-phenanthroline component are regularly employed in MOF synthesis.<sup>14-16</sup> Despite being a weakly emissive molecule, 1,10-phenanthroline can be built into linkers that exhibit emission bands, ranging from the UV to the near infrared region of the electromagnetic spectrum.<sup>17</sup> 2-(4-Carboxyphenyl)imidazo(4,5-*f*)-1,10-phenanthroline (HNCP) is an interesting derivative as it embodies two potential coordination domains, encouraging the formation of structures with higher dimensionalities (Scheme S1, ESI†).<sup>10,15</sup> Furthermore, the long  $\pi$ -conjugated linker can impart additional properties to frameworks in terms of coordinative strength and excellent photoabsorptivity.<sup>18</sup> Previously, we used HNCP to form a Ru<sup>II</sup>/Co<sup>II</sup> mixed-metal MOF (**PhotoMOF**) with large channel apertures, 2.1 nm in diameter, exhibiting high CO<sub>2</sub> adsorption and relatively long triplet <sup>3</sup>MLCT lifetimes.<sup>19</sup> To investigate the photoactive properties of this class of MOF without a Ru<sup>II</sup> photosensitizer, MOFs incorporating 2-(4-carboxy-1,1'-biphenyl)imidazo(4,5-*f*)(1,10)phenanthroline (HNCPP) were synthesised to access photoactive MOFs with larger pore apertures. These MOFs comprise ideal sites for CO<sub>2</sub> adsorption between interpenetrated nets and the ideal Qst and surface area properties for selective CO<sub>2</sub> sorption (*vide infra*).

HNCPP was synthesised by a Debus-Radziszewski reaction forming the imidazole backbone of the ligand (ESI†).<sup>20</sup> To accurately compare the lengths of the two linkers, the distance from the carbon centre on the carboxylate tail to the nitrogen phenanthroline atoms was measured from crystallographic data with HNCPP being 15.8 Å in length, while HNCP spans 11.5 Å.<sup>19</sup> [Mn(NCNP)<sub>2</sub>]<sub>n</sub> (**1**) was prepared by a solvothermal

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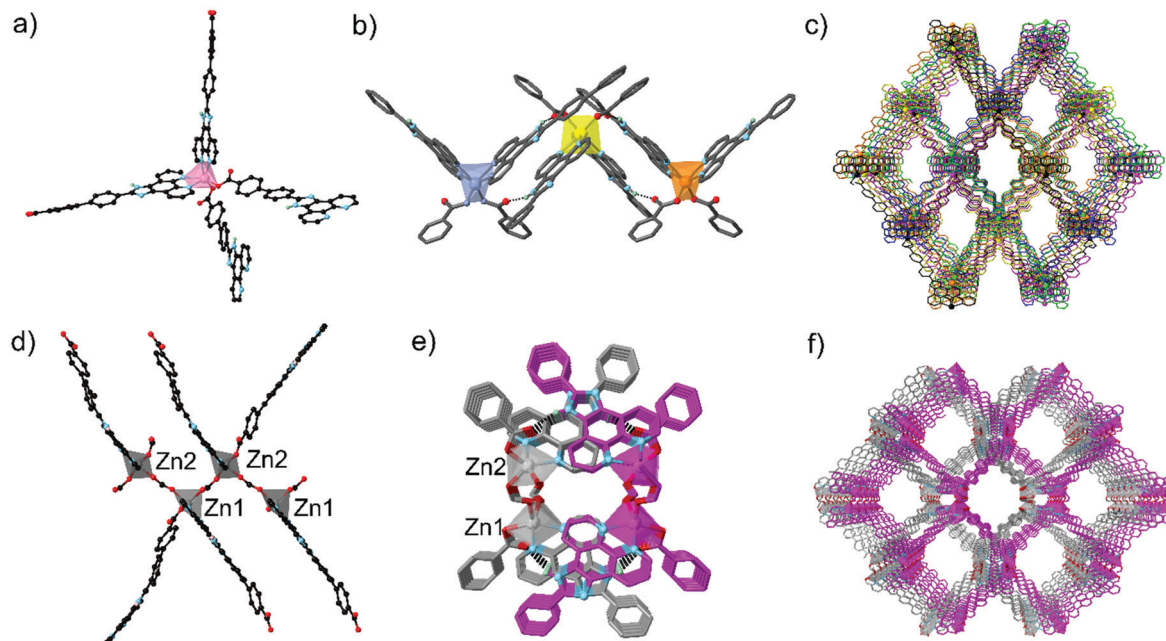
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† Electronic supplementary information (ESI) available: TGA, PXRD, isotherm data, isosteric heat of adsorption, BET calculations and DFT-D3 calculations. CCDC 2010335 and 2010336. For ESI and crystallographic data in CIF or other electronic format see DOI: 10.1039/d0cc05715k



**Fig. 1** (a and d) Coordination environment around the Mn centre of  $[\text{Mn}(\text{NCP}^{\ominus})_2]_n$  (**1**) and the two unique Zn centres of  $[\text{Zn}_2(\text{NCP}^{\ominus})(\kappa^1\text{-O}(\text{NCP}^{\ominus}))(\mu\text{-O}_2\text{CH})_2]_n$  (**2**), respectively; (b and e) imidazole to free carboxylate oxygen N–H...O hydrogen bonding between interpenetrated nets for **1** and **2**, respectively, viewed along the *a* axis for **1** and *b* axis for **2**; (c and f) the extended structures of **1** and **2** respectively, as viewed down the crystallographic *a* axes. The interpenetrated nets are identified through multiple colour assignments. Colour code: Mn: light pink (figure b); livid yellow and orange represent Mn centres from interpenetrating nets, Zn: grey, (figure e); grey and pink represent Zn centres from interpenetrating nets); black: carbon, lime: hydrogen, red: oxygen, blue: nitrogen.

reaction using  $\text{Mn}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$  and HNCPP, employing formic acid as a modulator, in DMF at 100 °C. Single crystal X-ray diffraction (SCXRD) of the isolated yellow needle crystals revealed that **1** crystallises in the orthorhombic space group *Fddd* with an asymmetric unit consisting of half a manganese atom and one  $\text{NCP}^{\ominus}$  ligand. The Mn centre adopts a distorted octahedral geometry, completed by four nitrogen donor atoms, from two chelating phenanthroline moieties, and two monodentate carboxylate–oxygen atoms from two different  $\text{NCP}^{\ominus}$  linkers (Fig. 1).

This coordination environment can be considered as pseudotetrahedral if each chelating phenanthroline moiety is considered as one connecting point. This configuration leads to a 3D diamondoid network, with the overall structure containing six such nets interpenetrating (Fig. S1 and S4, ESI<sup>†</sup>). Despite vast interpenetration, the structure contains large, 1D channels propagating down the *a* axis, giving pores of  $7.2 \times 12$  Å. Along each column of Mn clusters aligned parallel to the *a* axis, adjacent Mn centres are linked together by  $\text{NCP}^{\ominus}$  ligands with an Mn...Mn distance of 9.70 Å. Close  $\pi$ – $\pi$  and N–H... $\pi$  interactions exist between the phenanthroline moieties from adjacent units. In addition, close N–H...O hydrogen bonds (2.695(5) Å) between the NH from the imidazole ring and the uncoordinated carboxylate oxygen atom further promote network interpenetration (Table S2, ESI<sup>†</sup>). Phase purity of the bulk yellow crystals was confirmed *via* powder X-ray diffraction (PXRD) (Fig. S1, ESI<sup>†</sup>). The thermostability of **1** was established *via* thermogravimetric analysis (TGA) under  $\text{N}_2$  and revealed that the MOF exhibits three decomposition steps, at 180, 220 and 380 °C (Fig. S3, ESI<sup>†</sup>).

A solvothermal reaction between  $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ , HNCPP, and formic acid as modulator, in DMF at 100 °C, afforded yellow-orange needle crystals suitable for SCXRD. Structural analysis revealed the crystals to be  $[\text{Zn}_2(\text{NCP}^{\ominus})(\kappa^1\text{-O}(\text{NCP}^{\ominus}))(\mu\text{-O}_2\text{CH})_2]_n$  (**2**), crystallizing in the orthorhombic space group *Pna2*<sub>1</sub> with an asymmetric unit consisting of two  $\text{NCP}^{\ominus}$  ligands, two crystallographically distinct Zn ions and two formate bridges (Fig. 1). Each Zn is coordinated by one chelating phenanthroline moiety, one monodentate carboxylate and two formate ligands, giving an overall pseudotetrahedral geometry. Zigzag zinc formate chains propagate, down the *a* axis, with each unique zinc atom being connected to two others through two bridging formate linkers. These chains are then crosslinked by the  $\text{NCP}^{\ominus}$  ligands, creating an overall 3D network with 4-connected metal nodes and *zst*<sup>21</sup> topology (Fig. S4, ESI<sup>†</sup>).

There are close  $\pi$ – $\pi$  interactions (3.08–4.06 Å) between the phenanthroline and the phenylcarboxylate moieties on each zinc centre. The overall structure consists of two interpenetrated nets, which are held in place by close N–H...O hydrogen bonds (2.76–2.86 Å) between the uncoordinated carboxylate oxygen atom and the imidazole NH (Table S3, ESI<sup>†</sup>). *syn*-Parallel  $\pi$ – $\pi$  stacking between the imidazole groups from adjacent nets further stabilise the interpenetrated structure.<sup>22</sup>

**2** contains large 1D channels propagating along the *a* axis, with pore dimensions of  $15.3 \times 20.6$  Å and one edge lined with imidazole nitrogen atoms. Bulk phase purity was confirmed by PXRD (Fig. S2, ESI<sup>†</sup>) while TGA under  $\text{N}_2$  (Fig. S3, ESI<sup>†</sup>) reveals a two-step decomposition pathway at 220 and 325 °C.

The surface areas, total pore volumes and propensity for **1** and **2** to adsorb CO<sub>2</sub> was investigated using gas adsorption techniques. Activation of **1** was performed by washing the MOF with DMF, followed by cyclohexane exchange and evacuation at 80 °C. The Brunauer–Emmett–Teller (BET) surface area was determined to be 857 m<sup>2</sup> g<sup>-1</sup> using N<sub>2</sub> isothermal data recorded at 77 K (Fig. S5, ESI†). The isotherms also exhibit a small adsorption step at very low pressure (0.0035 *P/P*<sub>0</sub>), alluding to flexibility between interpenetrated nets in the structure (Fig. S6, ESI†).<sup>23</sup> A pore size of 11.8 Å was calculated from the N<sub>2</sub> adsorption isotherm at 77 K and *P/P*<sub>0</sub> = 0.995 (Fig. S10, ESI†). Activation of **2** was performed by washing the MOF in DMF, followed by DCM exchange and evacuation at 100 °C. The N<sub>2</sub> adsorption isotherm performed at 77 K revealed a BET surface area of 940 cm<sup>2</sup> g<sup>-1</sup> and a pore size of 12.7 Å (Fig. S7 and S11, ESI†).

Gas sorption capabilities were further investigated through recording CO<sub>2</sub> adsorption isotherms and calculating the enthalpy of adsorption using the Clausius–Clapeyron method. CO<sub>2</sub> uptake at 100 kPa at 273 K, 288 K and 298 K for **1** was 37 cm<sup>3</sup> g<sup>-1</sup>, 22.5 cm<sup>3</sup> g<sup>-1</sup> and 13 cm<sup>3</sup> g<sup>-1</sup>, respectively, resulting in a zero-coverage enthalpy of adsorption of 31 kJ mol<sup>-1</sup> (Fig. S8, ESI†). For **2**, CO<sub>2</sub> uptake at 273 K and 298 K was 17 and 12.5 cm<sup>3</sup> g<sup>-1</sup> respectively, giving a zero-coverage enthalpy of adsorption of 19 kJ mol<sup>-1</sup> (Fig. S9, ESI†). All CO<sub>2</sub> isotherms exhibited a typical type I shape and displayed an adsorption/desorption hysteresis, more pronounced at lower temperatures. We attribute this hysteresis to attractive interactions between the CO<sub>2</sub> guest and the conjugated phenyl systems lining the pore walls. Duplication experiments show that this hysteresis was not an effect of short equilibration times while running the experiments.

To further elucidate the presence of strong interactions between the CO<sub>2</sub> molecules and the framework atoms, density functional theory calculations, including dispersion corrections, (DFT-D3) were performed to calculate the binding energies and identify the most favourable adsorption site. Both MOFs revealed similar binding energies for CO<sub>2</sub> of ca. 31 kJ mol<sup>-1</sup>, which can be understood through close examination of the favourable adsorption site of each CO<sub>2</sub> in the framework. DFT-D3 calculations for **1** show this site to be in tight void spaces created by back-to-back carboxy-phenyl rings running along the *c* axis of the MOF (Fig. 2b), creating an overall largest cavity diameter (LCD) of 7.3 Å. Close π ··· O=C=O interactions are present between the NCPP phenyl rings and the guest CO<sub>2</sub> within the range of 2.77–3.64 Å. This denotes **1** as an exceptional candidate for CO<sub>2</sub> sorption from flue gas according to recent idealised criteria. This identifies materials with a *Q*<sub>st</sub> > 30 kJ mol<sup>-1</sup>, LCD of 5–7.5 Å and surface area < 1000 m<sup>2</sup> g<sup>-1</sup>.<sup>24</sup> DFT-D3 calculations for **2** show CO<sub>2</sub> to be nestled between the biphenyl NCPP tails of two interpenetrating nets, which have an LCD of 6.7 Å, causing close π ··· O=C=O interactions (Fig. S12, ESI†). These close interactions also pose **2** to be an excellent candidate for selective CO<sub>2</sub> sorption.

The employment of a photoactive linker prompted further investigation into the photophysical properties of **1** and **2** through examination using UV-visible absorption spectroscopy

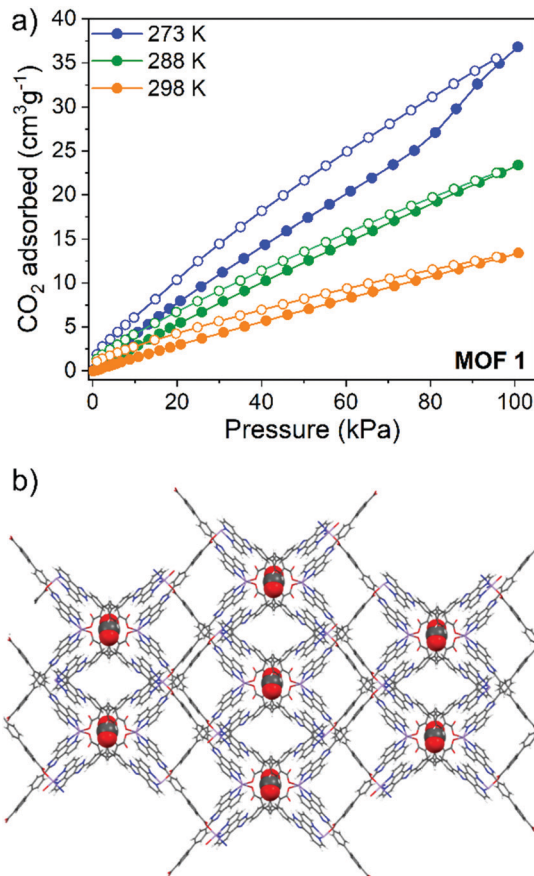


Fig. 2 (a) CO<sub>2</sub> adsorption (filled circles) and desorption (open circles) isotherms for **1** and (b) DFT-D3 optimization showing the preferential site of adsorption of CO<sub>2</sub> in **1**.

and time-correlated single photon counting (TCSPC). Due to the excellent stability of both **1** and **2** in DMF and the instability of **1** in moisture, measurements were made as suspensions in DMF. The absorption maxima ( $\lambda_{\text{max}}$ ) for the free HNCPP molecule was at 434 nm and assigned to an intra-ligand  $\pi \rightarrow \pi^*$  transition (Fig. 3a). The similar  $\lambda_{\text{max}}$  between HNCPP and the MOFs suggests that deprotonation and coordination to the metal centre has little effect on the excitation energy of the linker. Nevertheless, the coordination of NCPP<sup>-</sup> to Zn<sup>II</sup> and Mn<sup>II</sup> has slightly different effects on the absorption spectrum of the ligand. A slight red shift relative to the HNCPP maxima is observed in the absorption of **1** and **2** ( $\lambda_{\text{max}}$  = 443 and 439 nm, respectively). This can be attributed to the framework structure hindering  $\pi$ – $\pi$  stacking in **1** and **2**. The absorption spectra of **1** also shows an extra absorption peak at  $\lambda_{\text{max}}$  = 416 nm. This is ascribed to the LMCT between the NCPP<sup>-</sup> ligand and the Mn<sup>II</sup> centre based on previous research using HNCPP linkers.<sup>19</sup>

TCSPC decays for both HNCPP and **1** were recorded in DMF at  $\lambda_{\text{emis}}$  = 450 nm and highlight the reduced decay lifetime of **1**. **2** was not studied as the MOF included formate co-linkers and aggregated during the steady state PL measurement, therefore complicating a direct comparison with the HNCPP linker and **1** (Fig. 3b). Preliminary TCSPC experiments were first performed on the HNCPP ligand in DCM and DMF to accurately determine



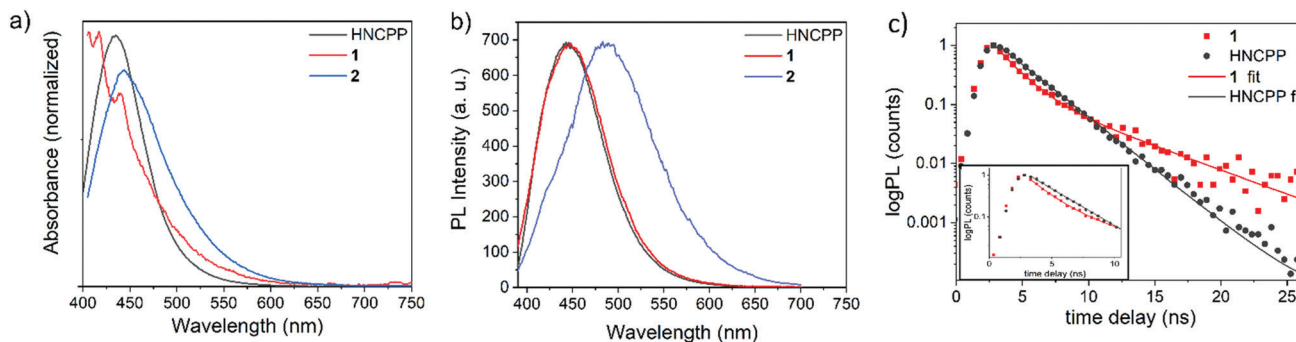


Fig. 3 Optical characterisation (a) absorption and (b) steady-state PL spectra of HNCPP, **1** and **2**. (c) Time-correlated single photon counting decays of HNCPP and **1**.

the correct solvent medium (Fig. S13, ESI<sup>†</sup>). In DCM, the HNCPP showed a rapid lifetime decay which was improved when the solution was diluted by a factor of 10. Nevertheless, DMF was selected as the solvent medium as DCM exhibited extensive noise levels due to ligand aggregation.

The DMF ligand solution shows a monoexponential decay with a lifetime of 2.44 ns (Fig. 3c). Photoluminescent (PL) decay, as measured by TCSPC, shows a quenching of NCPP lifetime upon coordination to Mn<sup>II</sup> ions in **1**. The quenching effect in the MOF indicated electron/energy transfer processes occurring throughout the framework structure. The PL lifetime of **1** is biexponential with decay lifetime components of 1.55 ns, attributed to the LMCT decay, followed by 5.30 ns. This longer, step of PL decay is a process with a magnitude much lower than the fast quenching component. This step could be due to aggregation of luminophores<sup>25</sup> or the formation of long-lived triplets due to intersystem crossing facilitated by the metal centres in the MOF.<sup>26</sup> To a first approximation, the PL lifetime gives a quenching of 35% compared to the PL lifetime of the ligand.

In conclusion, MOFs constructed from the HNCPP linker and Mn<sup>II</sup> or Zn<sup>II</sup> result in frameworks with large pore apertures and accessible 1D channels. Of significant interest are the photophysical properties of **1**, which exhibits rapid electron transfer between the NCPP<sup>-</sup> ligand and the Mn<sup>II</sup> centre. Both **1** and **2** display varying degrees of network interpenetration, providing a scaffold for strong interactions with CO<sub>2</sub> molecules. DFT-D3 investigations further highlight **1** and **2** as prime CO<sub>2</sub> selective candidates. These characteristics, combined with the photoactivity of **1**, frames future investigation into NCPP MOFs for photocatalytic CO<sub>2</sub> reduction.

We thank S. G. Telfer and C. Forsyth for their assistance. Part of this research was undertaken using the MX2 beamline at the Australian Synchrotron, part of ANSTO, and made use of the ACRF detector. R. B. acknowledges the National Computing Infrastructure and Pawsey supercomputing facilities for the Computational Resources. W. S. acknowledges SFI (13/IA/1896) and the ERC (Supramol CoG 2014-647719).

## Conflicts of interest

There are no conflicts to declare.

## Notes and references

- Z. Ji, H. Wang, S. Canossa, S. Wuttke and O. M. Yaghi, *Adv. Funct. Mater.*, 2020, 2000238.
- A. Schneemann, V. Bon, I. Schwedler, I. Senkovska, S. Kaskel and R. A. Fischer, *Chem. Soc. Rev.*, 2014, **43**, 6062.
- L. K. Macreadie, R. Babarao, C. J. Setter, S. J. Lee, O. T. Qazvini, A. J. Seeber, J. Tsanaktsidis, S. G. Telfer, S. R. Batten and M. R. Hill, *Angew. Chem., Int. Ed.*, 2020, **59**, 6090.
- L. K. Macreadie, E. J. Mensforth, R. Babarao, K. Konstas, S. G. Telfer, C. M. Doherty, J. Tsanaktsidis, S. R. Batten and M. R. Hill, *J. Am. Chem. Soc.*, 2019, **141**, 3828.
- G. Mínguez Espallargas and E. Coronado, *Chem. Soc. Rev.*, 2018, **47**, 533.
- W. P. Lustig, S. Mukherjee, N. D. Rudd, A. V. Desai, J. Li and S. K. Ghosh, *Chem. Soc. Rev.*, 2017, **46**, 3242.
- E. A. Dolgoplova, A. M. Rice, C. R. Martin and N. B. Shustova, *Chem. Soc. Rev.*, 2018, **47**, 4710.
- M.-L. Hu, S. A. A. Razavi, M. Piroozzadeh and A. Morsali, *Inorg. Chem. Front.*, 2020, **7**, 1598.
- T. Zhang and W. Lin, *Chem. Soc. Rev.*, 2014, **43**, 5982.
- Z.-H. Yan, M.-H. Du, J. Liu, S. Jin, C. Wang, G.-L. Zhuang, X.-J. Kong, L.-S. Long and L.-S. Zheng, *Nat. Commun.*, 2018, **9**, 3353.
- V. Stavila, A. A. Talin and M. D. Allendorf, *Chem. Soc. Rev.*, 2014, **43**, 5994.
- P. Samanta, S. Let, W. Mandal, S. Dutta and S. K. Ghosh, *Inorg. Chem. Front.*, 2020, **7**, 1801.
- H.-Q. Yin and X.-B. Yin, *Acc. Chem. Res.*, 2020, **53**, 485.
- S. Zhang, Y. Yang, Z.-Q. Xia, X.-Y. Liu, Q. Yang, Q. Wei, G. Xie, S.-P. Chen and S.-L. Gao, *Inorg. Chem.*, 2014, **53**, 10952.
- H.-Y. Sun, C.-B. Liu, Y. Cong, M.-H. Yu, H.-Y. Bai and G.-B. Che, *Inorg. Chem. Commun.*, 2013, **35**, 130.
- W. Xu, Y.-J. Tang, L.-Q. Zheng, J.-M. Xu, J.-Z. Wu, Y.-C. Ou and M.-L. Tong, *Inorg. Chem.*, 2019, **58**, 13766.
- G. Accorsi, A. Listorti, K. Yoosaf and N. Armaroli, *Chem. Soc. Rev.*, 2009, **38**, 1690.
- X. Zhou, J.-L. Peng, C.-Y. Wen, Z.-Y. Liu, X.-H. Wang, J.-Z. Wu and Y.-C. Ou, *CrystEngComm*, 2017, **19**, 6533.
- L. Martins, L. K. Macreadie, D. Sensharma, S. Vaesen, X. Zhang, J. J. Gough, M. O'Doherty, N.-Y. Zhu, M. Rütger, J. E. O'Brien, A. L. Bradley and W. Schmitt, *Chem. Commun.*, 2019, **55**, 5013.
- S. Saxer, C. Marestin, R. Mercier and J. Dupuy, *Polym. Chem.*, 2018, **9**, 1927.
- M. O'Keeffe, M. A. Peskov, S. J. Ramsden and O. M. Yaghi, *Acc. Chem. Res.*, 2008, **41**, 1782.
- S. Ray and A. Das, *J. Mol. Struct.*, 2015, **1089**, 146.
- X. Sun, Y. Ma, J. Zhao, D.-S. Li, G. Li, L. Zhang and Y. Liu, *Dalton Trans.*, 2018, **47**, 13158.
- C. Altintas, G. Avci, H. Daglar, A. Nemati Vesali Azar, S. Velioglu, I. Erucar and S. Keskin, *ACS Appl. Mater. Interfaces*, 2018, **10**, 17257.
- J. Mei, Y. Hong, J. W. Y. Lam, A. Qin, Y. Tang and B. Z. Tang, *Adv. Mater.*, 2014, **26**, 5429.
- J. Calbo, M. J. Golomb and A. Walsh, *J. Mater. Chem. A*, 2019, **7**, 16571.