

Cite this: *Chem. Sci.*, 2021, 12, 11936

# Perovskite indoor photovoltaics: opportunity and challenges

Kai-Li Wang,<sup>a</sup> Yu-Hang Zhou,<sup>a</sup> Yan-Hui Lou<sup>b</sup> and Zhao-Kui Wang<sup>id</sup>\*<sup>a</sup>

With the rapid development of the Internet of Things (IoT), photovoltaics (PVs) has a vast market supply gap of billion dollars. Moreover, it also puts forward new requirements for the development of indoor photovoltaic devices (IPVs). In recent years, PVs represented by organic photovoltaic cells (OPVs), silicon solar cells, dye-sensitized solar cells (DSSCs), etc. considered for use in IoT mechanisms have also been extensively investigated. However, there are few reports on the indoor applications of perovskite devices, even though it has the advantages of better performance. In fact, perovskite has the advantages of better bandgap adjustability, lower cost, and easier preparation of large-area on flexible substrates, compared with other types of IPVs. This review starts from the development status of IoTs and investigates the cost, technology, and future trends of IPVs. We believe that perovskite photovoltaics is more suitable for indoor applications and review some strategies for fabricating high-performance perovskite indoor photovoltaic devices (IPVs). Finally, we also put forward a perspective for the long-term development of perovskite IPVs.

Received 15th June 2021  
Accepted 4th August 2021

DOI: 10.1039/d1sc03251h

rsc.li/chemical-science

## 1. Introduction

In high speed under the background of modernization, indoor photovoltaics (IPVs) has attracted much attention with the emergence of Big Data and the Internet of Things (IoT), owing to the billions of product demand gap for the self-powered devices.<sup>1</sup> The IoTs refer to things that automatically communicate wirelessly between various electronic devices through the Internet. The digital information and real-time data can be collected or exchanged by sensors, terminal devices, health monitoring, *etc.*, which are widely used in intelligent buildings, radio-frequency identification (RFID) sensors, retail industries, and wearable devices.<sup>2–4</sup> In the new era of IoTs, radio has raised new requirements regarding size, weight, energy consumption, and cost reduction.<sup>5</sup> These applications are independent and off-the-grid, and it is wise and necessary to replace non-self-rechargeable batteries through power supplement *via* indoor photovoltaics.<sup>6</sup> In recent years, IPV as a device to convert indoor light to electric energy has been a powerful technology in the IoTs system.<sup>7–9</sup> For example, chalcogenide (*i.e.*, GaSe<sub>2</sub> and CuInGaSe<sub>2</sub>) solar cells,<sup>10,11</sup> dye-sensitized (DSSCs),<sup>12–14</sup> organic photovoltaic (OPVs),<sup>15,16</sup> and perovskite solar cells (PSCs)<sup>17,18</sup> have received significant attention in the IPVs field, owing to their advantages of simple fabrication technology, adjustable bandgap, and high performance. For optimizing IPVs, the

suitable absorption spectra, large open-circuit with low energy loss, minimized trap-mediated charge recombination and leakage currents, and excellent stability of indoor photovoltaic materials and devices are the key issues.<sup>19</sup> Furthermore, there always have been contradictions between the heavy, complex, and unsustainable portable power source and the portability of the IoTs system. The cumbersome power source becomes a burden for daily use and shortens the lifetime of mobile devices.<sup>20,21</sup> Hence, the development of simple, lightweight, low-cost, and self-powered devices is urgent for the IoTs system.

Notably, the PVs that are used to convert sunlight are not necessarily suitable for indoor applications due to the Sun's emission spectra and indoor light source. The emission spectra of indoor light sources (fluorescent lamps (FL) and light-emitting diodes (LED)) are located in the range of 200–700 nm and for solar AM1.5 spectrum is about 300–1100 nm. Meanwhile, the intensity of FL and LED are also three orders of magnitude lower than AM1.5.<sup>22,23</sup> Therefore, choosing a light-absorbing material with a suitable bandgap to absorb the indoor light source effectively is necessary. According to the theoretical simulations, the absorber materials with a 1.8–1.95 eV bandgap are ideal for high-performance IPVs.<sup>24–26</sup> This is the main reason that the high-performance solar cells with a bandgap from 1.1 to 1.6 eV (*e.g.*, crystalline silicon solar cells) designed for the one solar illumination intensity are unsuitable for indoor applications.<sup>27,28</sup> In addition, IPVs have higher requirements for defects and interface contact due to the too low indoor light source intensity that can only generate very few carriers. Park *et al.* demonstrated that doping chlorine into perovskite decreased the bulk defects of perovskite, which can

<sup>a</sup>Institute of Functional Nano & Soft Materials (FUNSOM), Jiangsu Key Laboratory for Carbon-Based Functional Materials & Devices, Soochow University, Suzhou, Jiangsu 215123, China. E-mail: zkwang@suda.edu.cn

<sup>b</sup>College of Energy, Soochow Institute for Energy and Materials Innovations, Soochow University, Suzhou 215006, China



suppress the ion migration and non-radiative recombination and improve the power density to  $3.525 (231.78) \mu\text{W cm}^{-2}$  under 400 lux.<sup>29</sup> Brown *et al.* also reported the role of interface defects on indoor flexible PSCs based on the  $\text{TiO}_2$  compact layer and the mesoporous layer, evidencing that the interface defects are crucial for the indoor performance of devices.<sup>30</sup>

In recent years, researchers have also tried to use various PV materials to absorb indoor light sources for fabricating IPVs. For example, DSSCs have been reported to perform well in indoor low-light sources. However, the high cost of this type of device limits the further development of its commercialization.<sup>31</sup> In contrast, PSCs have the advantages of efficient processing, lightweight, and low cost, making them have great potential in the IPVs field. Considering the different emission spectra of the light source, the absorber layer material of IPVs should have a medium bandgap and a narrow absorption spectrum in the visible light range to obtain good spectral matching.<sup>32,33</sup> Perovskite can freely adjust the bandgap without complicated synthetic routes and expensive preparation costs.<sup>34,35</sup> Furthermore, the bandgap of perovskite materials can freely change from 1.18 to 2.6 eV by a simple solution-processed component engineering, and its excellent light absorption ability and long carrier diffusion distance are also more suitable for low light conditions.<sup>36–38</sup>

This review summarizes the market demand, cost, application field, and market trend of IPVs. We also discuss the existing issues and predict the research direction of IPVs. The device performance of various types of IPVs will be compared. We believe that perovskite indoor photovoltaics (PIPVs) have the potential to lead the IPVs industry after solving the critical issues of stability and toxicity. Finally, we will be introducing the strategies for fabricating high-performance PIPVs.

## 2. Perspective of indoor photovoltaics

### 2.1 Application potential of IPVs

The IoTs system is referred to the automatic wireless communication between various electronic devices *via* the Internet, *i.e.*, real-time data collection and exchange of digital information by the electronic devices, and applied to smart home, aerospace, alarm systems, and indoor wearable technology. IPV as a mobile self-power device required in the IoTs system has diversification on the specifications, power, morphology, cost, and lighting environment to adapt to the different working environments. Therefore, IPVs have enormous potential to be used on various miniature mobile platforms, such as wearable electronic devices, wireless antennas, sensors, and biomedical treatments, as exhibited in Fig. 1a.<sup>39–41</sup> In addition, the influence of indoor light sources must also be considered. Normally, a simulated solar source AM1.5 ( $100 \text{ mW cm}^{-2}$ ) is commonly used to evaluate the performance of PVs in the laboratory.<sup>42,43</sup> However, adequate sunlight may not always be met due to the reference of weather, environment, time, *etc.* Thus, the performance of PVs under artificial indoor lighting sources, such as LED, CFL, halogen lamps, and incandescent bulbs with special color

appearance and different intensities (Fig. 1b) also needs to be considered.<sup>44</sup> In the near future, light energy from the environment through IPVs for continuous power supply is necessary.<sup>45,46</sup> Fig. 1c shows an example of indoor wireless interconnection equipment with the power provided by IPVs ( $1 \text{ cm}^2$ ). For example, organic based IPVs can achieve a good PCE of about 30% (corresponding input power of  $0.521 \text{ mW cm}^{-2}$  and output power of  $15 \mu\text{W}$ ), which is enough to drive many electronic applications (RTID tags, wristwatches, calculators and standby devices).<sup>39,47</sup> It is worth noting that the PCE of PSCs under indoor conditions is mostly above 35%.<sup>48–50</sup> That means the same area of PIPVs can drive more electronic devices, and it will also become an excellent candidate in the IPV field.

PVs have been combined with watches, calculators, and sensors for many years (Fig. 1d), owing to the stable power output and the excellent performance under low-light sources.<sup>45,51</sup> In addition, IPVs show great potential to create a huge market for indoor renewable energy. For example, some companies such as WSL Solar,<sup>52</sup> Powerfilm,<sup>53</sup> and Soelms<sup>54</sup> are commercializing amorphous Si modules as IPVs. GCEl<sup>55</sup> and Ricoh<sup>56</sup> are also using DSSCs as IPVs for commercialization. Particularly, amorphous Si and dye-based PVs can even drive simple devices, such as watches, calculators, and sensors at a light intensity of 20 lux.<sup>54</sup> According to forecasts, hundreds of millions of wireless sensors will be installed indoors in the next ten years.<sup>55</sup> It can be said that IPVs have a good development prospect even if there are still many difficulties. Among them, the greatest difficulty is the long-term stability of devices and the toxicity of materials (such as cadmium (Cd), lead (Pb), and arsenide (As)). Over the years, the manufacture of photovoltaic devices in the research and manufacturing sector has received great attention, although it has been banned in some areas because of its toxicity.<sup>57–60</sup> In fact, Alta company recently closed its IPVs production due to the lack of suitable investors.<sup>61</sup> This suggests that IPVs manufacturing technology must be developed rapidly in order to meet the requirements of long-term human development. In addition, it is also crucial that the IPVs can be fabricated on a flexible and large-area substrate to meet the different market needs. Photovoltaic panels can be installed directly into these types of equipment or simply installed on panels on walls, floors, or other surfaces, to nearby power equipment or return power to the building's power supply system.

In today's fast modernization, IPVs have also been rapidly developed. However, the commercialization of IPVs is still laggard compared with the development of solar modules. For example, reports show that the global market value of IPVs is only \$140 million,<sup>45</sup> which is far below the \$100 billion of solar energy modules.<sup>62</sup> We all know that in order for something new to replace something that already exists, it must have advantages that the body (human) cannot refuse, such as cost and performance. Typically, the cost of the production model is dependent on the production scale of the PV system.<sup>46,63</sup> However, in order to fabricate low-cost IPVs, using inexpensive materials and an excellent technique process is far-sighted. This strategy is beneficial to the growth of the expected market size of different types of IPVs, which is the function of the price per





Fig. 1 (a) The equipment obtains energy through IPVs power generation under indoor lighting conditions. Reproduced from ref. 39 with permission from the [Wiley-VCH], copyright [2018]. (b) Emission spectra of different light sources. Reproduced from ref. 45 with permission from the [Elsevier], copyright [2019]. (c) Powered by IPVs of 1 square centimeter under indoor lighting and the corresponding drivable electronic equipment. Reproduced from ref. 39 with permission from the [Wiley-VCH], copyright [2018]. (d) The cost and market size of IPVs as a function of time (years). (e) Expected market size of different photovoltaic applications (2018–2027). (f) Manufacturing cost and annual production scale and indoor usage of different types of IPVs, (g) forecast of IPVs and WS market size, and (h) market demand for photovoltaic cells/modules in different energy demand fields (2016–2023). Reproduced from ref. 45 with permission from the [Elsevier], copyright [2019].

watt of different equipment, as shown in Fig. 1e.<sup>45,64</sup> Each sphere shows the forecast of the market size (horizontal span) and price per watt (vertical span) from 2018 to 2027. In the next ten years (2018–2027), PVs will be mainly used in integrated buildings (approximately two orders of magnitude increase), and its price per watt will also have a great price advantage in the possible equipment. To realize the vision of the IoTs ecosystem, *i.e.* billions of sensor nodes and wireless antenna connected to the network, relying on a single node to reduce power usage is the most effective method. In recent years, the research trend of energy efficiency and low power hardware protocols is to solve this problem, improve energy efficiency to reduce latency, and improve data reliability.<sup>65,66</sup> Fig. 1f shows the comparison of the production of system units based on low-volume amorphous Si, III–V, PSCs, CdTe, and Si compounds

with the annual production volume. Among them, OPV, PSCs, and CdTe based cells have lower cost.<sup>45,66</sup> In addition, within the range of 5 years, the OPV devices show lower costs if using the R2R method.

PV cells used as indoor wireless node power supply systems can promote the growth of the IPV market. The current global market for IPV cells is \$140 million.<sup>45,67</sup> The estimated size of the wireless sensor (WS) and IPVs market in the next ten years (2017–2023) is billions of dollars (Fig. 1g).<sup>45</sup> The IPV market will experience a spurt of development in the next ten years, reaching \$850 million in 2023 and hopefully exceed a billion dollars in the following years. By 2023, the market demand for photovoltaic energy is expected to reach 60 million units per year.<sup>67</sup> In addition, Fig. 1h shows the forecast for the next 8 years (2018–2023). IPVs not only form a part of the building



integration PVs (BIPV) to obtain a higher market value ratio, but also continue to exhibit a higher market ratio relative to PVs used for remote sensing and telecommunications.<sup>45</sup> This means that IPVs will become an integral part of the power ecosystem for the Internet of Things. In addition, the global IPV market is expected to grow to \$5.1 billion in 2023, representing an annual growth rate of 33.6 percent.<sup>68</sup>

## 2.2 Research progress of IPVs

When there is no solar radiation, IPVs mainly collect the energy emitted by the artificial light source (CFL, LED, incandescent, halogen). The illumination intensity is usually three orders of magnitude lower than the sunlight. The difference in the light source directly leads to the difference in the spectral range and the bandgap of the matching absorber. Fig. 2a shows the maximum limit PCEs of PVs under different indoor light sources, according to the detailed equilibrium limit theory and Shockley and Queisser.<sup>24,69</sup> This calculation is performed based on the assumption that the above bandgap energy of all absorbed under different lighting conditions, and the carrier was completely extracted. We can find that the optimal bandgap for AM1.5 is *ca.* 1.34 eV, and for the indoor light source (LED and CFL) is *ca.* 1.9 eV. Surprisingly, the maximum theoretical efficiency of 1.9 eV bandgap material can be close to 60% when IPVs are under the 3-color LED condition. However, under the

condition of AM1.5, the calculated efficiency limit of PVs with a bandgap of 1.1 eV does not exceed 40%. This is due to the light absorption range of the absorber with a bandgap of 1.9 eV almost perfectly covering the indoor light source spectrum (200–700 nm), making the photons be fully utilized by active materials under indoor light sources with a narrow spectrum. Meanwhile, compared with sunlight, indoor light sources can achieve higher efficiency due to the narrower spectral band reducing the transparency loss and thermal loss associated with the broadband solar spectrum.<sup>70</sup> This is the reason for silicon-based solar cells having lower PCE under the indoor lighting conditions compared with OPVs or PSCs.<sup>71–74</sup>

In addition, it is challenging to accurately estimate indoor performance since there is no recognized standard for indoor spectral quality and irradiance (*e.g.*, AM1.5 for the outdoor standard). Lin *et al.* summarized the existing indoor light intensity measurement defects by lux meter, which essentially explained the irregularity of the existing test standards.<sup>75</sup> Thus, a universal, standard, and accurate indoor spectroscopy method and equipment are necessary. In recent years, OPVs, DSSCs, and GaAs based IPVs have been developed rapidly. CFL and LED are commonly used as indoor light sources to evaluate the performance of IPVs. As shown in Fig. 2b, Peters *et al.* used a detailed efficiency balance limit method to calculate the theoretical maximum efficiency of IPVs (*i.e.*, Si, GaAs, OPV, and perovskite based IPV) under 1 W m<sup>-2</sup> of CFL and LED lighting.<sup>45</sup>



Fig. 2 (a) The bandgap of the light-absorbing layer material under different light sources and the maximum photoelectric conversion efficiency of the corresponding IPV. Reproduced from ref. 24 with permission from the [Institute of Electrical and Electronics Engineers Inc], copyright [2015]. (b) The statistics of maximum efficiency (under CFL and LED illumination) and bandgap for IPVs, so far. Reproduced from ref. 45 with permission from the [Elsevier], copyright [2018]. (c) The PCE and  $V_{OC}$  distribution of various types IPVs under the indoor light source. Reproduced from ref. 96 with permission from the [Royal Society of Chemistry], copyright [2019]. (d) OPVs maximum power output density and  $V_{max}/V_{OC}$  ratio under indoor light source. Reproduced from ref. 77 with permission from the [American Institute of Physics], copyright [2016].



The PCE of perovskite-based IPVs is higher than the others. Notably, the maximum PCE values of IPVs can reach an astonishing 52% under the indoor light source of CFL and LED illuminations. Currently, the efficiency of IPVs is around 40%. Notably, it cannot obtain a higher  $V_{oc}$  due to the mismatch of the optical bandgap (<2 eV).

Silicon, a dominant cell material, and the Si-based solar cells with record solar PCE is over 26% under AM1.5. However, this type of PVs demonstrates an incompatible PCE of 8% under the indoor light source.<sup>76–80</sup> To overcome the limitation of the bandgap of crystalline silicon in IPVs, amorphous silicon (a-Si, the bandgap is wider, 1.6 eV) has become one of the leading technologies in indoor photovoltaics. Shieh *et al.* reported a p-a-SiC:H window layer that was used to enhance the efficiency of amorphous Si-based solar cells to 25.56  $\mu\text{W cm}^{-2}$  under 500 lux.<sup>81</sup> In addition, CdTe and CIGS are the most successful photovoltaic materials than silicon. However, CIGS has a low shunt resistance under low light conditions, resulting in a significant decrease in the efficiency of devices as the light intensity decreases.<sup>82</sup> In contrast, CdTe (with a bandgap of 1.5 eV) based IPVs can maintain high performance under diffuse radiation and indoor light sources, which could have promoted its establishment of a strong commercial advantage in the photovoltaic market.<sup>83</sup> However, a relatively small amount of public data indicates that CdTe-based IPVs are only 10.9% efficient under CFL lighting conditions, which is not enough to advance the CdTe-based IPVs.<sup>84</sup> Analogously,  $\text{Cu}_2\text{ZnSn}(\text{S},\text{Se})_4$  (CZTSSe) thin film cells also have efficiencies approaching 10% under CFL and AM1.5G spectra.<sup>85</sup> Among the existing IPVs, GaAs based solar cells are extremely competitive, and their efficiency exceeds 20% on the flexible substrates under low light indoor sources.<sup>86,87</sup> Initially, some materials were used to optimize the bandgap close to 2 eV, such as  $\text{Ga}_x\text{In}_{1-x}\text{P}$  and  $\text{Al}_x\text{GaAs}$  (1.8–1.9 eV) to obtain a higher efficiency under LF or LDE. However, the experimental results showed similar efficiency as GaAs, GaInP, and AlGaAs-based IPVs under an indoor light source.<sup>86,88,89</sup> OPVs, a high-performance PV with adjustable bandgap and high absorption for visible light region spectrum, has been developed rapidly in recent years. The best efficiency of OPVs has reached 17.8% with AM1.5 illuminations,<sup>90</sup> and more than 28% efficiency was obtained under an indoor light source (1000 lux) using materials with a bandgap of 1.8 eV.<sup>91</sup> Furthermore, dye-sensitized IPVs also shows excellent performance under low light conditions, and optimal efficiency of 31.8% was demonstrated under CFL illuminance (1000 lux).<sup>92,93</sup> Recently, Freitag *et al.* reported more than 34% efficiency under 1000 lux for DSSCs based on a copper(n/i) electrolyte, and the large-area (16  $\text{cm}^2$ ) devices were used to power machine learning on wireless nodes.<sup>94</sup> Very recently, Grätzel *et al.* reported the highest efficiency of 34.5% under 1000 lux for DSSCs with copper(n/i)-based electrolyte, indicating that this type of photovoltaics still occupies a leading position in the field of IPVs.<sup>95</sup>

In 2019, Liu *et al.* summarized the PCE,  $V_{oc}$ , and bandgap of various types of IPVs, such as Si thin-film, perovskite, GaAs, InGaP, DSSCs, and OPVs based IPVs (Fig. 2c).<sup>96–102</sup> As the optical bandgap increases (1.15–2 eV), corresponding IPVs often show

higher  $V_{oc}$  and PCE under indoor light source illumination. It is worth mentioning that PSCs have an indoor PCE of close to 40% (corresponding to an optical bandgap of only 1.55 eV) due to their excellent light absorption and defect tolerance. Regarding the performance of IPVs under different indoor light sources, it is necessary to consider the effects of different materials on  $J_{sc}$ ,  $V_{oc}$ , and FF (Fig. 2d).<sup>77,96</sup> Taking OPVs as an example,  $J_{sc}$  can be related to the absorption of the donor material, which depends on the energy gap. Normally, in most PVs,  $J_{sc}$  is approximately linearly proportional to the intensity of the incident light.<sup>103–105</sup> The  $V_{oc}$  is a parameter related to light intensity ( $V_{oc} \sim \frac{nKT}{e} \ln\left(\frac{I_{ph}}{I_0}\right)$ ).<sup>106–108</sup> Then, the  $\Delta V$  between indoor and outdoor illumination can be estimated as  $\Delta V = \frac{nKT}{e} \ln\left(\frac{I_{ph,Sun}}{I_{ph,indoor}}\right)$ . Thus,  $V_{oc}$  will increase with the increasing light intensity, which explains the additional loss ( $\sim 0.2$  eV) of PVs under indoor light conditions.<sup>109,110</sup> In addition, for all systems, FF increases as the light intensity decreases. Under indoor lighting conditions, the FF of each system will be significantly higher than that under the AM1.5 condition.<sup>111,112</sup>

### 2.3 The accurate measurement of indoor photovoltaic

There is a clear difference between the indoor and outdoor photovoltaic performance of PVs. It is essential to evaluate the photovoltaic performance of IPVs accurately. Various indoor light sources are used in daily life, including indirect sunlight, fluorescent lamp, incandescent lamp, halogen lamp, LED bulbs, and other light sources with low light intensity (200–1000 lux). Normally, the indoor light intensity is described as illumination (lux) because the human eye is sensitive to the light spectrum. So, the illumination of the indoor light source can be converted using the corresponding incident power as equation:<sup>113</sup>

$$E_v = K_m \times \int_0^\infty E_\lambda \times V(\lambda) d\lambda$$

where the  $V(\lambda)$  is CIE spectral luminosity factor for human photopic vision and  $K_m$  is equal to 683  $\text{lm W}^{-1}$ .

In addition, accurate light source calibration and test conditions are the keys to measure the performance of IPVs accurately. Wong *et al.* proposed calibrating indoor light sources using a general LED lux meter with NIST-traceable calibration (*e.g.*, Extech LT40 NIST) and recommended that the maximum power point  $P_{max}$  and PCE values be used to measure the device performance accurately.<sup>113</sup> Very recently, Hou *et al.* proposed that the accurate test of IPVs should include the following parts. (1) The time instability of the light source for indoor PV measurement should be less than 2%. PV test should be carried out in the illumination area where the spatial distribution of light intensity is less than 2%. (2) Use a mask with the same or larger size than the transparent substrate of the cells to test the photoelectric conversion efficiency. (3) The spectrometer is more suitable for indoor light source calibration than the lux meter. Meanwhile, they also suggested that the comparison difference between  $J_{sc}$  (EQE) and  $J_{sc}(I-V) < 5\%$  is



suitable for verifying the accuracy of the test.<sup>114</sup> In general, in order to accurately measure the indoor photovoltaic performance, it is necessary to use a spectrometer to calibrate the light source and select the appropriate light source. Meanwhile, the light source parameters (such as light intensity, lux, *etc.*) and power parameters should also be included in the device performance description.

### 3. Research direction of IPVs

Like outdoor photovoltaics, the efficiency, life (stability), and cost of IPVs are the key to measuring the technical feasibility of the commercialization of photovoltaic devices (Fig. 3a).<sup>35,115</sup> In addition, large-area, flexible and modular IPVs play

a particularly important role in the industrialization of IPVs. This section will summarize the development direction of IPVs from these parts.

#### 3.1 Low-cost and stable IPVs

The comprehensive level of IPVs, *i.e.*, cost, efficiency, and stability, determines whether it can occupy a prominent position in today's competitive market of emerging photovoltaic technology.<sup>116</sup> Furthermore, the overall cost of IPVs is determined by individual costs and lifetime, reducing the individual costs, enhancing the lifetime of single devices, increasing the effective output time, and reducing the maintenance costs. At present, the new generation thin film PVS represented by single



Fig. 3 (a) The golden triangle of solar cell development and (b) comparison of perovskite solar cells and silicon solar cells. Reproduced from ref. 35 with permission from the [Nature Publishing Group], copyright [2018]. (c) The picture of a flexible solar cell and the *J*-*V* curves of the best-performing devices under 400 lux under the LED light. Reproduced from ref. 30 with permission from the [Tsinghua University Press], copyright [2018]. (d) Rigid PSCs module. Reproduced from ref. 76 with permission from the [Nature Publishing Group], copyright [2018]. (e) The schematic diagram of perovskite large-area modules. Reproduced from ref. 136 with permission from the [Nature Publishing Group], copyright [2018].



perovskite cells has high PCE (more than 40% under 1000 lux) and low manufacturing cost (reach half of crystalline Si).<sup>117,118</sup> However, the longest life (stability) of perovskite PV is estimated to be only one year (Fig. 3b).<sup>35</sup> It is not enough to change the fact that the inherent lifetime of thin-film IPVs is much lower than silicon-based cells (~25 years), even if considering the indoor working environment has relatively stable temperature and humidity. Further investigation reveals that most life expectancy is reported only for PV samples, and these samples do not meet the current industry standards.<sup>119</sup> Thus, further improvement of the lifetime of thin film-based IPVs is necessary for their industrialization. Meanwhile, we also notice that the most reported lifetime-related investigation is conducted on un-encapsulated samples, which do not meet current industry standards.<sup>120–122</sup> Furthermore, the encapsulation technique can effectively improve the lifetime while maintaining the performance of PVs. In fact, the encapsulation of PVs must be considered for real applications, even if its cost is estimated to exceed 60% of the total cost of PVs.<sup>123</sup> Therefore, reducing the packaging cost is a necessary condition for the industrialization of IPVs, and in-depth research on cheap materials and packaging technology is of great importance. In addition, different levels of water vapor transmittance are considered, *i.e.*, glass (zero) and flexible barriers ( $10^{-1}$  to  $10^{-6}$  g per m<sup>2</sup> per day), an excellent encapsulation technology, and packing materials are critical to the lifetime of IPVs.

### 3.2 Flexible IPVs

For the IoTs system, it is better for IPVs to be flexible to match the multi-scene applications. Flexible IPVs can be combined on mobile platforms, *i.e.*, buildings, watches, backpacks, and portable power supplies. Currently, indium tin oxide (ITO) film is widely used in the PV industry. However, the high processing temperature, brittleness, and manufacturing cost of ITO films limit their applications in wear-resistant, durable, and flexible IPVs.<sup>124</sup> Therefore, the development of flexible substrates or electrode materials with high conductivity, high transmittance, durability, and lightness has a great effect on the IPVs industrialization. Furthermore, the mechanical flexibility of electrodes must be considered because IPVs usually have curved shapes or flexible features. Considering the lower light intensity of the indoor light source, the substrate or transparent conducting electrode (TCE) must be highly transparent and have a smooth surface, which allows the maximum number of photons to pass through and minimizes the recombination center to avoid current leakage. Currently, flexible electronics, *i.e.*, graphene, conductive polymers, and silver nanowires, have been widely introduced into IPVs.<sup>125–127</sup> Notably, as one of the most commonly used flexible substrate materials, PET is widely used in flexible PIPVs. Until now, the vast majority of flexible PVs are fabricated on this substrate. In 2016, Brown *et al.* reported that the flexible PIPVs based on CH<sub>3</sub>NH<sub>3</sub>PbI<sub>3-x</sub>Cl<sub>x</sub> reached 10.8% and 12.1% efficiencies under 200 lux and 400 lux lighting conditions, respectively (Fig. 3c).<sup>30</sup> Park *et al.* demonstrated that butyl acetate (BA) replaces the CB-treated perovskite films during spin coating, which can improve the gradual

nucleation and the growth of grains. The optimal devices exhibit a remarkable maximum power density of 0.063 mW cm<sup>-2</sup> and efficiency of 23.33% under 400 lux.<sup>128</sup> Brown *et al.* fabricated flexible perovskite modules and cells based on a combination of SnO<sub>2</sub> and mesoporous-TiO<sub>2</sub>. The devices of a mesoporous TiO<sub>2</sub> scaffold layer over SnO<sub>2</sub> exhibited better indoor performance. For example, the cells have maximum power densities of 9.77 and 19.2 μW cm<sup>-2</sup> under 200 and 400 lux illuminations, respectively.<sup>129</sup> In 2020, Brown *et al.* reported flexible perovskite solar cells based on the roll-to-roll ITO-coated ultra-thin flexible glass (FG) substrates. The optimized devices based on FG incorporated a mesoporous scaffold over SnO<sub>2</sub> compact layers and obtained a record efficiency of 20.6% (16.7 μW cm<sup>-2</sup>) and 22.6% (35.0 μW cm<sup>-2</sup>) under 200 and 400 lux LED illumination, respectively.<sup>48</sup> It is believed that in the near future, there will be more and more researches and reports on flexible PIPVs. In addition, some reports on ITO-free flexible PVs can promote the further development of flexible PIPVs due to their good stretchability and flexibility.<sup>130–133</sup>

### 3.3 Large-area IPVs

Since the light intensity of the indoor light source is ten times lower than AM1.5, the production of large-area IPVs can effectively meet the requirement of various power equipment by increasing the output power of a single cell. The output power of single-cell PVs depends on the device type (*e.g.*, OPVs, DSSCs, and CuInGaSe). The output power of single-cell PVs with a small area (<1 cm<sup>2</sup>) has also been significantly improved in recent years. Meanwhile, many techniques like laser scribing and screen printing for the fabrication of the functional layer have been developed.<sup>134,135</sup> As shown in Fig. 3d, the large-area fabrication of PV modules is inseparable from the application of these techniques.<sup>76</sup> At the same time, in order to meet the requirements of more application platforms, it is necessary for IPVs to develop a higher output voltage or current by connecting individual cells in series with other techniques (such as laser etching). As shown in Fig. 3e, the fabrication of IPV modules requires multiple etching (at least three times), and the accuracy, width, and position of the slit will directly determine the overall output of the module.<sup>136</sup>

## 4. Challenges and opportunities of PIPVs

### 4.1 Overview of PIPVs

Perovskite is a magical photovoltaic material in the photovoltaic field in recent years. PSCs have many advantages, such as low cost, efficiency, and large easy area fabrications, which are expected to become the mainstay of the photovoltaic industry in the future. The certified efficiency in the laboratory has approached 25.5% under AM1.5 radiation.<sup>137–143</sup> Recently, these types of cells have shown high indoor PCE exceeding 35%.<sup>144–147</sup> It is worth mentioning that perovskite benefits from its excellent properties by repeatedly exhibiting higher PCE under the same bandgap conditions. Furthermore, the advantages of the easy solution process, large area, and flexibility are very



important to drive PIPVs on the road to industrialization. Meanwhile, perovskite materials ( $\text{CsSnI}_3$ ,  $\text{MAPbI}_3$ ,  $\text{MAFAPbI}_3$ ,  $\text{CsPbI}_3$ , and  $\text{CsPbBrI}_2$ , etc.) show variable bandgap (1.18–2.6 eV), enabling them to be fabricated as high-performance PIPVs for the specific indoor light source. Undeniably, there are still many difficulties on the road to the commercialization of PIPVs. Although the leakage of lead can be effectively reduced through physical encapsulating, it cannot essentially reduce the use of lead.<sup>148–150</sup> Number of PSCs based companies have recently focused on developing related products but have still not applied these in the IoTs market. Fortunately, the recent rise of tin-based perovskite solar cells has completely replaced the use of the toxic element lead with tin.<sup>151–153</sup> This may be the new hope for PIPV to move towards industrialization.

#### 4.2 Wide range of bandgap

Perovskite materials have a wide range of adjustable forbidden bandwidths, making them particularly attractive in multi-junction tandem solar cells and IPVs. Many reviews have summarized the perovskite materials and developments of single cells and tandem PSCs.<sup>154–159</sup> In this section, we aim to overview the adjustable bandgap of perovskite materials and their application in PIPVs. Metal halides perovskite crystal is described as an  $\text{ABX}_3$  structure, where A is a monovalent cation (MA, FA, Cs, and Rb), B is a kind of divalent metal cation (Pb, Sn, and Ge), and X is a halide anion (I, Br, and Cl).<sup>160</sup> The application of PSCs began with methylammonium lead iodide ( $\text{MAPbI}_3$ ) with a bandgap of 1.55 eV,<sup>161,162</sup> developed to replace cations and anions and expanded to other perovskite compositions.<sup>163</sup> As shown in Fig. 4a, the bandgap of the perovskite film can be continuously adjusted from infrared (1.15 eV) to ultraviolet (up to 3 eV) by the composition engineering of mixed cations or anions.<sup>164</sup> Perovskite material has a wide range of continuous bandgap adjustment ability, so it is an ideal candidate for IPVs, and can also be used for the preparation of PVs in special environments (e.g., microwaves and infrared).

The device performance of PVs is the basis for the fabrication of high-performance PIPVs, which determines the output power of single-cell and multi-cell PIPVs. For example, the highest theoretical PCE limit of perovskite is 31%, which is far higher than the laboratory record efficiency (~25%).<sup>165</sup> Therefore, selecting perovskite materials with a matching bandgap from the existing mature PSCs to fabrication of PIPVs can avoid the performance degradation caused by the technology level. At present, the high-performance perovskite materials  $\text{CsPbBrI}_2$  (1.89 eV) and  $\text{CsPbI}_3$  (1.7 eV) can be used to fabricate high-performance PIPVs, considering the matching of bandgap and the indoor light source spectra (200–700 nm).<sup>166,167</sup> To be sure, more and more perovskite materials (replace cations and anions) can be used to fabricate high-performance PIPVs with the development of PV technology.

#### 4.3 Low-cost PIPV module

To achieve a low-level energy cost compared with traditional energy, PIPVs must have the comprehensive advantages of low cost, high performance, and long-term stability.<sup>168</sup> We can

estimate the cost of PIPVs from the cost of PSCs because the cells or modules show an almost identical structure. In recent years, PSCs have not only demonstrated high-performance under AM1.5 (PCE ~ 25%), but also provided effective strategies to reduce the manufacturing cost by solution process based on low-temperature deposition process and roll-to-roll manufacturing on flexible substrates.<sup>167–169</sup> Huang *et al.* placed the perovskite precursor solution (ink) on the preheated substrate (70–145 °C), crystallizing it into a black solid perovskite film during the solvent evaporation.<sup>168</sup> The successful application of large-area manufacturing strategies has reduced the fabrication cost. Moreover, according to the preliminary life-cycle assessment result, perovskite solar modules are expected to have a lower environmental impact and shorter energy return time (EPBT) indicators.<sup>170–173</sup> The production cost of perovskite-based photovoltaics is expected to be further reduced in the future, owing to the increase in the continuous operating time of PVs, which can effectively reduce the replacement and maintenance cost of PVs modules.

The cost of perovskite modules and accessories must also be considered for commercial production. Some groups have evaluated the manufacturing cost of perovskite modules with different geometries and have proposed related estimates of LCOE.<sup>118,174,175</sup> Han *et al.* concluded that perovskite photovoltaic modules could be produced with simplified module geometry at an ultra-cost of \$30–41 per square meter (excluding back-sealing glass, junction boxes, or wires). According to their assumptions, the LCOE value can be as low as 3.5–4.9 cents per kilowatt per hour, which is much lower than the cost of fossil fuel.<sup>174</sup> Subsequently, Egan *et al.* conducted a complete evaluation, including the use of expensive metals (e.g., Ag and Au) and vacuum deposition technology (e.g., thermal evaporation), and estimated that the PSC manufacturing cost and LCOE value range from \$87–140 per  $\text{m}^2$  and 9.0–18.6 cents per kilowatt per hour, respectively.<sup>175</sup> In 2017, Heben *et al.* re-evaluated the economic potential of PSCs by developing a bottom-up cost model of perovskite photovoltaic modules (feasible low-cost materials and manufacturing processes) (Fig. 4b).<sup>115</sup> They believed that PSCs could play a role as a cost leader in the PV field if the key remaining issues can be resolved.<sup>118</sup> Furthermore, this estimated cost will be further reduced with large-area processes (e.g., roll-to-roll (R2R) and screen printing), low-cost transport layers, or electrode materials that are widely used in PIPV.

#### 4.4 Large-area and flexible PIPVs

The large-area preparation process of PIPV cells or modules can refer to the large-area PSCs. At present, large-area technologies (e.g., slot-die coating, roll-to-roll, and inkjet printing) have widely been used to manufacture flexible and rigid PSCs or modules. Spin-coating is mainly used to fabricate small cells (about 0.1  $\text{cm}^2$ ) and larger devices with an area of 1  $\text{cm}^2$  because this method can easily control the chemical composition and thickness of the film. The spin-coating method of the perovskite film includes one-step and two-step spin-coating processes.<sup>176</sup> Two-step spin-coating method, that is, first spin-coating lead



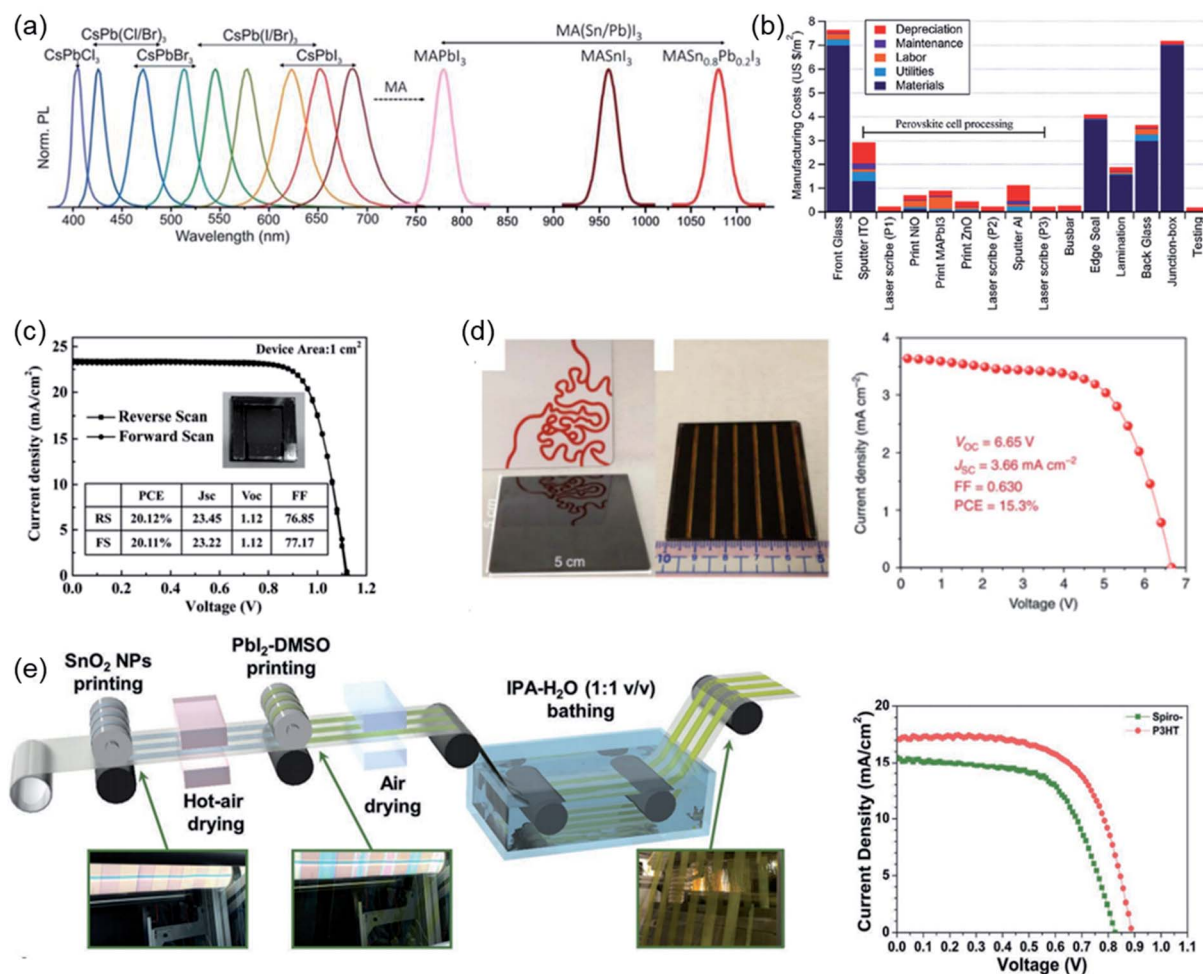


Fig. 4 (a) Different component perovskite materials and corresponding PL peaks. Reproduced from ref. 164 with permission from the [John Wiley and Sons Ltd], copyright [2019]. (b) Statistics on the cost of perovskite photovoltaic modules. Reproduced from ref. 115 with permission from the [Royal Society of Chemistry], copyright [2017]. (c) The photo and  $J$ - $V$  curve of PSCs (area is 1 cm<sup>2</sup>). Reproduced from ref. 177 with permission from the [Nature Publishing Group], copyright [2017].<sup>177</sup> (d) The photograph of 5 × 5 cm MAPbI<sub>3</sub> films and  $J$ - $V$  curves of PSCs (active area is 12.0 cm<sup>2</sup>). Reproduced from ref. 182 with permission from the [Nature Publishing Group], copyright [2018]. (e) Schematic diagram of PSCs fabricated by the R2R process and the corresponding  $J$ - $V$  curve. Reproduced from ref. 185 with permission from the [Wiley-VCH], copyright [2014].<sup>185</sup>

iodide, then spin-coating MAI dissolved in a solvent such as IPA, and finally annealing to form a large-area perovskite film is widely used to fabricate large-area perovskite films. As shown in Fig. 4c, You *et al.* demonstrated a perovskite film with an area of 1 cm<sup>2</sup> by a two-step spin coating method and obtained a PCE of 20.1% after controlling the residual PbI<sub>2</sub> on the perovskite surface.<sup>177</sup> However, the material utilization rate of the spin coating process is too low and difficult to form a pinhole-free, uniform perovskite film on a large area of the substrate (>1 cm<sup>2</sup>). Thus, it is imperative to apply large-area techniques (*e.g.*, blade coating,<sup>162</sup> slot-die coatings,<sup>178</sup> spray coating,<sup>179</sup> ink-jet printing,<sup>180</sup> and roll-to-roll<sup>181</sup>) to fabricate large-area perovskite films. Qi *et al.* used MAI to let the reaction of lead hydrogen triiodide (HPbI<sub>3</sub>(Cl)) and CH<sub>3</sub>NH<sub>2</sub> gas occur by a quick gas-solid reaction, obtaining a high-quality perovskite film with a thickness of more than one μm (5 × 5 cm), which delivered a PCE up to 15% for solar modules (Fig. 4d).<sup>182</sup> For

PIPVs, low-intensity indoor light conditions require that the perovskite film could be fabricated in a large area to increase the output current and module production.

Furthermore, considering the characteristics of portability, wearability, and durability of IoTs system, a flexible manufacturing process based on a large area is also necessary for PIPVs. Compared with other perovskite films fabrication techniques, the continuous R2R process has the advantages of fast fabrication speed and low cost. Gao *et al.* demonstrated the blow-assisted pouring method (BADC) and NH<sub>4</sub>Cl additive to assist the preparation of MACH<sub>3</sub>NH<sub>3</sub>PbI<sub>3</sub> film in the air on a flexible substrate with an area of 6.25 cm<sup>2</sup> by R2R, yielding a PCE of 11.16%.<sup>183</sup> In 2019, Kim *et al.* demonstrated the R2R process to fabricate PSCs in the air and achieved a PCE of 11.7%.<sup>184</sup> In the same year, Seo *et al.* applied gravure printing to fabricate flexible devices for the first time. The perovskite layer was prepared by the two-step method with partial R2R



processing, and the corresponding devices exhibited a PCE of 9.7% (Fig. 4e).<sup>185</sup> The application of the R2R method in the fabrication of perovskite films can effectively reduce the production cost. However, the shortcomings of low PCE limit its development in the future. Furthermore, the flexible large-area PIPVs also require the development of advanced techniques, such as interface engineering, component engineering, and solvent atmosphere engineering, to improve the PCE of the devices.

#### 4.5 The challenge of PIPV

Although PSCs have advantages in synthesis route and cost, their poor stability under humidity, heat, ultraviolet radiation, and oxygen conditions limit their real application.<sup>186</sup> Normally, the stability of perovskite depends on its structure configuration, such as the  $\text{MX}_6^{4+}$  octahedral structure and the connection between the A cation and the adjacent octahedron. Hence, adjusting the elements and ingredients can greatly improve the intrinsic stability of the perovskite structure.<sup>187</sup> Karunadasa *et al.* synthesized a 2D  $(\text{PEA})_2(\text{CH}_3\text{NH}_3)_2[\text{Pb}_3\text{I}_{10}]$  perovskite, which demonstrates a significantly better humid stability than  $\text{MAPbI}_3$  perovskite.<sup>188</sup> In addition, interfacial modification or the use of a stable transport layer is another effective strategy to improve the air stability. Yang *et al.* demonstrated a different self-assembly monolayer (SAM) on the  $\text{SnO}_2$  transport layer to effectively improve carrier transport and air stability of the device.<sup>189</sup> Notably, all-inorganic ( $\text{CsPbX}_3$ ) perovskite has excellent humid and thermal stability due to the use of  $\text{Cs}^{2+}$  instead of  $\text{MA}^+$  to avoid irreversible decomposition of organic functional groups under light and heat conditions. In fact, although the stability of perovskite photovoltaic devices has been greatly improved, it is still far from reaching commercial requirements. It is worth noting that IPVs have a more moderate working environment than outdoor photovoltaics, which will greatly extend the stability of indoor photovoltaic devices. Undisputedly, it is necessary to further improve the lifetime of perovskite photovoltaics.

The use of lead, a heavy metal element in perovskite materials, has always been a concern. On the one hand, in order to eliminate the toxicity of perovskite materials, the use of tin, silver, bismuth, indium, copper, *etc.*, instead of lead for fabricating PSCs has also been well developed in recent years.<sup>190–194</sup> Chu *et al.* used a Lewis base indacenodithiophene-based organic acceptor (ITIC) to improve the morphology and optoelectronic properties of  $\text{Cs}_3\text{Sb}_2\text{I}_9$ -based solar cells. The optimal  $\text{Cs}_3\text{Sb}_2\text{I}_9/\text{ITIC}$  heterostructure-based devices obtained a PCE of 9.2% under 1000 lux illumination.<sup>192</sup> Furthermore, tin-based PIPVs have also been prepared and achieved excellent efficiency. Wang *et al.* demonstrated that incorporation of catechin into the  $\text{FA}_{0.75}\text{MA}_{0.25}\text{SnI}_2\text{Br}$  perovskite film suppresses oxidation. It is the first report on the indoor photovoltaic efficiency of tin-based perovskite solar cells, reaching 12.81% (1000 lux).<sup>195</sup> On the other hand, lead can be recycled to improve the sustainability of perovskite photovoltaics. Jung *et al.* reported the lead management process of iron-containing hydroxyapatite, which improved the electrostatic interaction induced Pb

adsorption through surface charge delocalization. The researchers purified the non-aqueous solvent containing Pb to meet the standards of the U.S. Environmental Protection Agency and recovered 99.97% of Pb ions through the formation of lead iodide.<sup>196</sup> However, although the research of lead recovery is still relatively few, it is at least an effective auxiliary measure to solve the problem of perovskite industrialization. Furthermore, with the application of encapsulation technology of photovoltaics becoming more and more mature, the toxicity of perovskite will be solved in the foreseeable future.

## 5. Technology of PIPV

Among the devices that collect a variety of energy (*e.g.*, light, heat, radio-frequency signals, and motion) to power IoTs systems, PVs are considered the ideal candidate, owing to their consistent availability, high output voltage, and power density.<sup>197</sup> Generally, the intensity of indoor light sources (CFL and LED) is about three orders of magnitude lower than the outdoor AM1.5G solar irradiation. The illuminance of the common indoor light source (FL and LED) is located in the range of 200–700 lux (corresponding irradiance is approximately 50–300  $\mu\text{W cm}^{-2}$ ).<sup>198–200</sup> The strategies of component engineering, interface engineering, and effective transporting materials are considered to be developed for fabricating high-performance PIPVs. In this section, we will summarize the high-performance PIPVs fabricated based on these strategies.

### 5.1 Composition engineering for PIPVs

Wide bandgap perovskite materials are considered for fabricating high-performance PIPVs due to the narrow spectral range (200–700 nm) of the indoor light source. However, the working environment of high-performance PSCs under AM1.5 (300–1000 nm) requires the bandgap of perovskite to be as small as possible to expand the light absorption range. Thus, the fabrication of high-performance PIPVs must be considered to adjust the bandgap of perovskite materials by composition engineering. For example, for the perovskites with  $\text{ABX}_3$  cubic crystal structure, the doping or replacement of the B and X sites is considered to change their bandgaps.<sup>201</sup> Furthermore, composition engineering can also improve the quality of perovskite film and reduce or eliminate the use of the toxic element Pb.

In recent years, composition engineering strategies have been widely used to fabricate high-performance PIPVs. For example, Wang's group doped lead oxalate ( $\text{PbC}_2\text{O}_4$ ) into perovskite precursor solution and realized the replacement of anions during the film annealing process (Fig. 5a).<sup>201</sup> This strategy improved the quality of the perovskite film by delaying the crystallization rate and then obtained a PIPV with a PCE of 34.86% under 1000 lux illumination. According to a previous report, the doping or replacement of the X site is considered to be the most effective method of bandgap adjustment. Feng *et al.* specifically designed triple-anion based  $\text{CH}_3\text{NH}_3\text{PbI}_{2-x}\text{BrCl}_x$  perovskite for indoor light collection. The PIPVs based on this kind of perovskite presented PCEs of 36.2% (0.1  $\text{cm}^2$ , 1000 lux)





Fig. 5 (a) Schematic diagram of PSCs fabricated with PbC<sub>2</sub>O<sub>4</sub> doping and the *J*-*V* curves of with/without PbC<sub>2</sub>O<sub>4</sub>-based devices. Reproduced from ref. 201 with permission from the [American Chemical Society], copyright [2020]. (b) The bandgap of MAPbI<sub>3</sub>, MAPbI<sub>2</sub>Br, and MAPbI<sub>2-x</sub>Br<sub>x</sub> perovskite and *J*-*V* curves of PSCs under indoor light conditions. Reproduced from ref. 202 with permission from the [Wiley-VCH], copyright [2019]. (c) The crystal structure of BiOI and Cs<sub>3</sub>Sb<sub>2</sub>Cl<sub>x</sub>I<sub>9-x</sub> and the *J*-*V* curves of the corresponding device under FL and WLED illuminations. Reproduced from ref. 205 with permission from the [Wiley-VCH], copyright [2020].

and 30% (2.25 cm<sup>2</sup>, 1000 lux), respectively (Fig. 5b).<sup>202</sup> In 2019, Chen *et al.* used a similar strategy to fabricate MA<sub>0.85</sub>CS<sub>0.15</sub>Pb(I<sub>1-x</sub>Br<sub>x</sub>)<sub>3</sub>-based PIPVs. They found that the PIPVs with 15% Br<sup>-</sup> ions exhibited higher indoor PCE of 26.4% under 1000 lux.<sup>203</sup> Analogously, Shim *et al.* reported that the 10% Br<sup>-</sup> doping could realize perovskite films with better crystallization, and the corresponding PIPVs showed a higher indoor PCE of 34.5 ± 1.2% (LED @ 1000 lux).<sup>204</sup> Particularly, lead-free perovskites-inspired materials (PIMs) have similar electronic structures as lead halide perovskites, but do not have the limitation of toxicity. Considering the environmental issues of IPVs, lead-free PIPVs must be developed in the future. In 2020, BiOI and Cs<sub>3</sub>Sb<sub>2</sub>Cl<sub>x</sub>I<sub>9-x</sub> based PIPVs were investigated by Pecunia *et al.* (Fig. 5c). They found that the BiOI and Cs<sub>3</sub>Sb<sub>2</sub>Cl<sub>x</sub>I<sub>9-x</sub> devices only presented a PCE of 1% under AM1.5. However, this PCE can be increased to 4–5% under indoor light.<sup>205</sup> In addition, the rapid development of tin-based PSCs in recent years also provides new candidates for eco-friendly lead-free IPVs.<sup>206,207</sup>

## 5.2 Interface engineering for PIPVs

In recent years, planar structure PSCs have attracted much attention owing to the simple fabrication process and low cost.<sup>208</sup> However, interface conditions, such as interface contact,

interface morphology, and interface defect, significantly affect the device performance of planar PSCs since these planar structure devices are very sensitive to the interfaces.<sup>209,210</sup> Interfaces in PSCs (*i.e.*, electron-transporting layer (ETL) or hole-transporting layer (HTL)/perovskite layer interface, front and back contact interfaces) strongly affect the charge extraction, recombination, and collection process.<sup>211</sup> For high-performance PIPVs, the Shockley-Read-Hall (SRH) trap assisted recombination at the interface limits the improvement of the device performance. The non-radiative energy loss, energy-level mismatch, and optical losses at the interfaces are fatal to the performance of PIPVs under low light intensity indoor sources. Interface engineering is an effective strategy to improve the interface and interlayer properties to overcome interface loss. Notably, interface engineering can also largely improve cell stability.<sup>212</sup>

In addition, interface engineering can effectively passivate defects on the perovskite film, modify the interface contact, and improve the carrier extraction/transport process with the goal of fabricating high-performance PIPVs. In 2018, Wang's group demonstrated high-performance MAPbI<sub>3</sub>-based IPVs with ionic liquid of (BMIM)BF<sub>4</sub> as a modification layer, which can pave the interface contact (PCBM/perovskite layer) and passivate the trap



states in the inverted, improving the electron transporting and extracting processes. The corresponding devices exhibit a recorded PCE of 35.20% under FL (1000 lux) (Fig. 6a).<sup>213</sup> Recently, Wang's group reported a lycopene modification layer to fabricate  $(\text{CsFAMA})\text{Pb}(\text{I}_{1-x}\text{Br}_x)_3$  films with fewer defects and better environmental stability, yielding an indoor PCE of 40.24% under LED (2700 K @ 1000 lux), which is the highest indoor efficiency among all PV systems till date.<sup>214</sup> Considering the narrow spectrum of indoor light sources (200–700 nm), suitable bandgap ( $\sim 1.8$  eV) of perovskite materials is more conducive to the fabrication of PIPVs with high performance and high  $V_{oc}$ . Thus, a series of wide bandgap perovskite materials (e.g.,  $\text{CsPbBrI}_2$  and  $(\text{FA}_{0.6}\text{MA}_{0.4})_{0.9}\text{Cs}_{0.1}\text{Pb}(\text{I}_{0.6}\text{Br}_{0.4})_3$ ) are considered for the fabrication of high-performance PIPVs. For example, Wang's group used  $(\text{NH}_4)_2\text{C}_2\text{O}_4 \cdot \text{H}_2\text{O}$  to treat  $\text{CsPbBrI}_2$  (1.89 eV) film to make high quality perovskite film with micrometer-scale and low trap density (Fig. 6b). The optimal devices presented a best indoor PCE of 28.48% under an FL of 1000 lux.<sup>215</sup> In 2020, Jen *et al.* also reported a simple strategy by applying PEACl, PEAI, and PEABr to reduce the energy loss and suppress the phase segregation of  $(\text{FA}_{0.6}\text{MA}_{0.4})_{0.9}\text{Cs}_{0.1}\text{Pb}(\text{I}_{0.6}\text{Br}_{0.4})_3$ -based (1.75 eV) PIPVs (Fig. 6c). The optimized

PIPVs by PEACl presented a PCE of 35.6% with  $V_{oc}$  of 1.08 V under 1000 lux.<sup>216</sup>

Similar to PSCs, PIPVs also consists of a perovskite layer sandwiched between two transporting layers (HTL and ETL) to form a multilayer structure. Considering the low light intensity of the indoor light source, the transporting layer with excellent properties can effectively reduce the charge non-radiative recombination loss at the interface and improve the extraction and transport process of carriers. On the one hand, the transporting materials for PSCs can also be considered for the fabrication of high-performance PIPVs. On the other hand, the different working environments of PIPVs and PSCs may also lead to the fact that charge transporting layers are not necessarily universal. For example, Tsoi *et al.* simulated different hole-transporting layers, such as mesoporous PPV (mPPV), carbon-based PPV (cPPV), and inverted PPV (iPPV) for PIPVs under an indoor environment (Fig. 7a). They found that the performance of mPPV-based indoor photovoltaic devices fabricated by Spiro-OMeTAD as the HTL were significantly better than other HTLs (PTAA, PEDOT:PSS and Poly-TPD), and the optimal device presented a maximum power exceeding  $111 \mu\text{W cm}^{-2}$ .<sup>217</sup> In 2019, Samuel *et al.* demonstrated NiO nanoparticle



Fig. 6 (a) Inverted structure PSCs, molecular structure of [BMIM]BF<sub>4</sub> and BCP and the  $J-V$  curves. Reproduced from ref. 213 with permission from the [Wiley-VCH], copyright [2018]. (b) Device structure of  $\text{CsPbBrI}_2$  based PSCs, molecular structure of  $(\text{NH}_4)_2\text{C}_2\text{O}_4$ , and the  $J-V$  curves of devices with/without treatment. Reproduced from ref. 215 with permission from the [Elsevier], copyright [2020]. (c) Schematic diagram of PIPVs, the chemical structures of PEACl, PEABr, PEAI, and the  $J-V$  curves of the device with/without treatment under CFL and LED irradiation. Reproduced from ref. 216 with permission from the [Elsevier], copyright [2020].



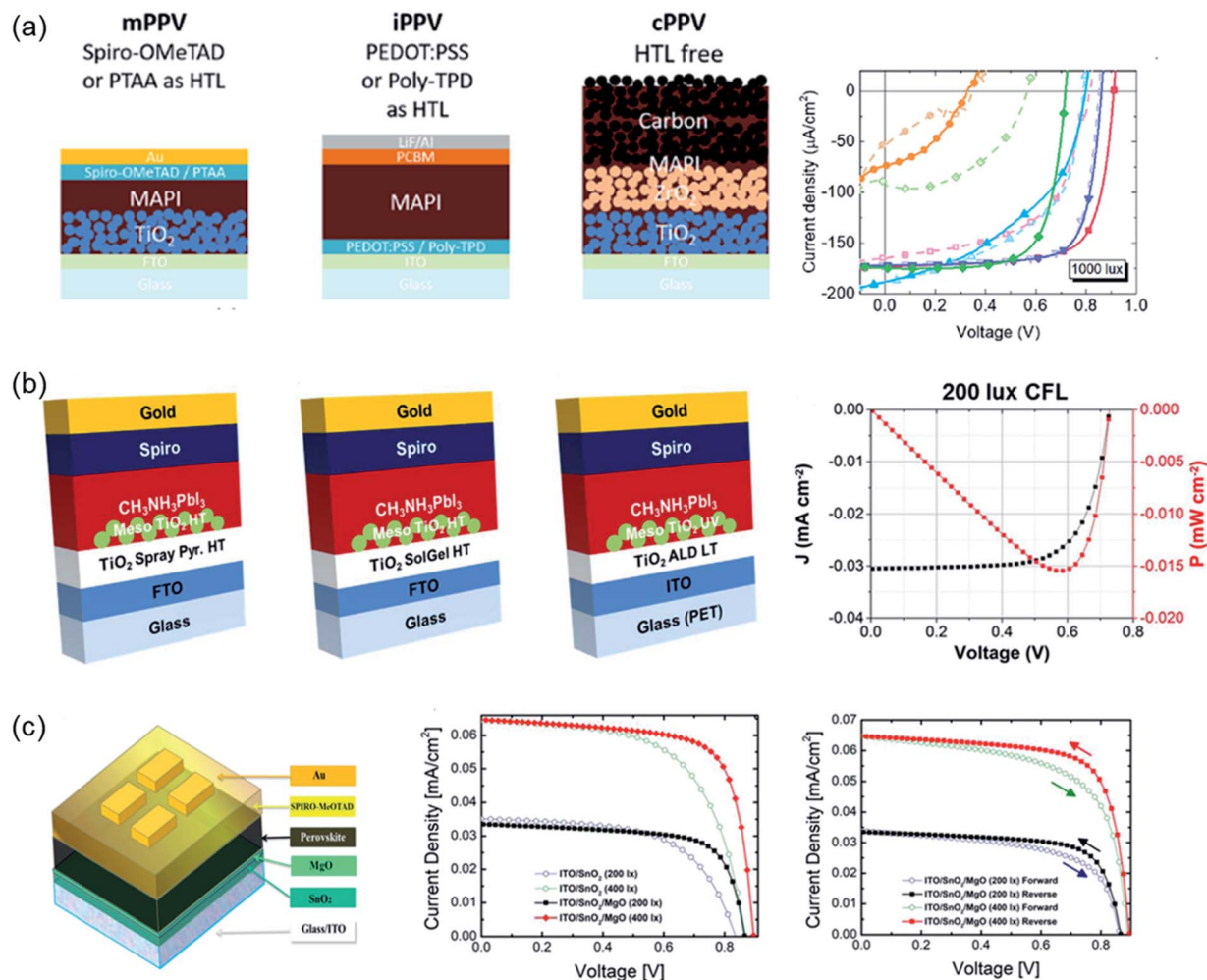


Fig. 7 (a) The structural diagram of the three perovskite devices, and the  $J$ - $V$  curve of the corresponding devices are at 1000 lux illumination. Reproduced from ref. 217 with permission from the [Wiley-VCH], copyright [2018]. (b) Schematic diagram of the structure of mesoporous TiO<sub>2</sub> prepared by different methods and the  $J$ - $V$  curve of the optimal device. Reproduced with permission from the [Elsevier], copyright [2016]. (c) Schematic diagram of the device structure fabricated by SnO<sub>2</sub> with MgO treatment and the  $J$ - $V$  curves of the corresponding devices under different light conditions. Reproduced from ref. 227 with permission from the [Elsevier], copyright [2018].

based film ( $\sim 6$  nm) as the HTL for MAPbI<sub>3</sub>-based PIPVs. The device presented a higher indoor PCE of 23% under FL illumination.<sup>218</sup>

In recent years, metal oxides (TiO<sub>2</sub>, InO<sub>3</sub>, ZnO, SnO<sub>2</sub>, etc.) have also been developed as ETL to fabricate high-performance inverted PSCs.<sup>219</sup> Brown *et al.* explored the effect of TiO<sub>2</sub> prepared by spray pyrolysis technique (SP), sol-gel process (SG) and atomic layer deposition (ALD) on the performance of MAPbI<sub>3</sub>-based PIPVs under FL illumination (Fig. 7b).<sup>220-223</sup> The PIPVs fabricated by ALD TiO<sub>2</sub> exhibited outstanding performance (PCE = 24%) under indoor light illumination (200 lux).<sup>223</sup> Very recently, SnO<sub>2</sub> with higher electron mobility, higher optical transparency, and wider bandgap has been developed to fabricate high-performance PIPVs.<sup>224-226</sup> Brown *et al.* introduced a thin layer of MgO over the SnO<sub>2</sub> to reduce the charge recombination at the interface (SnO<sub>2</sub>/perovskite layer) and improve the performance of PIPVs (Fig. 7c). The SnO<sub>2</sub>/MgO based PIPVs showed better power density values of 20.2  $\mu\text{W cm}^{-2}$  (200 lux)

and 41.6  $\mu\text{W cm}^{-2}$  (400 lux).<sup>227</sup> With the rapid development of material technology, we believe that more excellent charge-transporting materials will be designed and developed for the preparation of high-performance PIPVs.

## 6. Conclusion and perspectives

Under the background of the rapid development of IoTs and the gap in the demand for IPVs with hundreds of millions of markets, OPVs, DSSCs, and PIPVs have ushered in developing opportunities. Notably, the IPVs/PIPVs can not only be used in small devices such as wearable electronics, but also can be used as the main energy supply in some special situations (as BIPV) or under conditions of extensive coverage. Its maintenance cost will also be much lower than that of outdoor photovoltaics due to its milder operating environment. Furthermore, thin-film PVs represented by OPVs and DSSCs will occupy a dominant position in the current market due to their excellent



performance and low cost. Nevertheless, PIPVs will dominate the market in the near future owing to their advantages of high efficiency, low cost, large area, and high defect tolerance. The development trend of PIPVs is also like other IPVs, tending to be a large area, low cost, high stability, and flexible fabrication. Thus, this review summarizes the cost, performance, merits, and future development trends of various types IPVs, especially programing the development trend of PIPVs, and summarizes the strategies for high-performance PIPV. We believe that the fabrication and development of high-performance PIPVs should follow the following parts:

(A) Using composition engineering to adjust the bandgap and quality of perovskite films by optimizing the bandgap to 1.8–1.95 eV that fits the indoor light source.

(B) Using interface engineering to optimize crystallinity and passivate defects of perovskite films to reduce carrier non-recombination loss. Meanwhile, it is also important to optimize the transport layer material and minimize the recombination of the interfaces and the electrodes.

(C) Using high mobility, high flatness, and low-cost transporting materials to improve the extraction and transport process of carriers.

(D) Using ITO-free flexible substrates to fabricate high-performance PIPVs is more conducive to the advancement of their commercialization process due to their good stretch and flexibility.

(E) Using mature encapsulation techniques to improve the device lifetime.

Notably, considering the use of PIPVs devices in IoTs, the reduction in the use of the toxic element Pb (replaced by Sn, Rb, etc.) is often accompanied by a reduction in device performance. In addition, even considering the relatively mild indoor environment, the device lifetime of PIPV is still not enough to support its industrial development. It is undeniable that PIPV still has the potential to guide the IPVs in the future, owing to their excellent indoor performance.

## Author contributions

K. L. W. collect reference, organize images and write manuscripts; Y. H. Z. modify the manuscript and participate in discussions; all the authors discussed the results and commented on the manuscript. Y. H. L. and Z.-K. W. supervised the project.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

The authors acknowledge financial support from the National Natural Science Foundation of China (No. 62075148, 52073197) and the Natural Science Foundation of Jiangsu Province (No. BK20201413). This project was also funded by the Collaborative Innovation Center of Suzhou Nano Science and Technology,

and by the “111” Project of the State Administration of Foreign Experts Affairs of China.

## Notes and references

- 1 *The Internet of Things*, ed. D. Giusto, A. Iera, G. Morabito and L. Atzori, Springer, 2010, ISBN: 978-1-4419-1673-0.
- 2 H. Jin, T.-P. Huynh and H. Haick, *Nano Lett.*, 2016, **16**, 4194–4202.
- 3 M. Xie, K. Hisano, M. Zhu, T. Toyoshi, M. Pan, S. Okada, O. Tsutsumi, S. Kawamura and C. Bowen, *Adv. Mater. Technol.*, 2019, **4**, 1800626.
- 4 E. Ahmed, I. Yaqoob, I. A. T. Hashem, I. Khan, A. I. A. Ahmed, M. Imran and A. V. Vasilakos, *Computer Networks*, 2017, **129**, 459–471.
- 5 S. Li, L. Da Xu and S. Zhao, *Information Systems Frontiers*, 2015, **17**, 243–259.
- 6 I. Mathews, S. N. Kantareddy, T. Buonassisi and I. M. Peters, *Joule*, 2019, **3**, 1415–1426.
- 7 Y. Cui, Y. Wang, J. Bergqvist, H. Yao, Y. Xu, B. Gao and J. Hou, *Nat. Energy*, 2019, **4**, 768–775.
- 8 M. Freitag, J. Teuscher, Y. Saygili, X. Zhang, F. Giordano, P. Liska and A. Hagfeldt, *Nat. Photonics*, 2017, **11**, 372–378.
- 9 J. K. W. Ho, H. Yin and S. K. So, *J. Mater. Chem. A*, 2020, **8**, 1717–1723.
- 10 I. Repins, M. A. Contreras, B. Egaas, C. DeHart, J. Scharf, C. L. Perkins, B. To and R. Noufi, *Photovoltaics*, 2008, **16**, 235–239.
- 11 K. L. Chopra, P. D. Paulson and V. Dutta, *Prog. Photovoltaics Res. Appl.*, 2004, **12**, 69–92.
- 12 D. Song, M. Li, Y. Li, X. Zhao, B. Jiang and Y. Jiang, *ACS Appl. Mater. Interfaces*, 2014, **6**, 10.
- 13 M. Freitag, J. Teuscher, Y. Saygili, X. Zhang, F. Giordano, P. Liska, J. Hua, S. M. Zakeeruddin, J.-E. Moser, M. Gratzel and A. Hagfeldt, *Nat. Photonics*, 2017, **11**, 372.
- 14 S. Mathew, A. Yella, P. Gao, R. Humphry-Baker, B. F. E. Curchod, N. Ashari-Astani, I. Tavernelli, U. Rothlisberger, M. K. Nazeeruddin and M. Gratzel, *Nat. Chem.*, 2014, **6**, 242.
- 15 N. Gasparini, A. Salleo, I. McCulloch and D. Bara, *Nat. Rev. Mater.*, 2019, **4**, 229.
- 16 J. Zhao, Y. Li, G. Yang, K. Jiang, H. Lin, H. Ade, W. Ma and H. Yan, *Nat. Energy*, 2016, **1**, 15027.
- 17 J. Bisquert and E. J. Juarez-Perez, *J. Phys. Chem. Lett.*, 2019, **10**, 5889.
- 18 M. Saliba, T. Matsui, K. Domanski, J.-Y. Seo, A. Ummadisingu, S. M. Zakeeruddin, J.-P. Correa-Baena, W. R. Tress, A. Abate, A. Hagfeldt and M. Grätzel, *Science*, 2016, **354**, 206.
- 19 H. S. Ryu, S. Y. Park, T. H. Lee, J. Y. Kim and H. Y. Woo, *Nanoscale*, 2020, **12**, 5792–5804.
- 20 Z. L. Wang, *Adv. Mater.*, 2012, **24**, 279.
- 21 Z. L. Wang and W. Wu, *Angew. Chem., Int. Ed.*, 2012, **51**, 11700–11721.
- 22 B. Minnaert and P. Veelaert, *Thin Solid Films*, 2011, **519**, 7537–7540.



- 23 P. Yongqiang, B. Tao and H. Lingxia, *Infrared and Laser Engineering*, 2012, **41**, 2484–2488.
- 24 A. S. Teran, J. Wong, W. Lim, G. Kim, Y. Lee, D. Blaauw and J. D. Phillips, *IEEE Trans. Electron Devices*, 2015, **62**, 2170–2175.
- 25 A. Abbas, A. Abdollahinia, A. G. Aberle, S. Adachi, J. Adams, J. G. J. Adams and D. Aiken, *IEEE J. Photovoltaics*, 2013, **3**, 1465.
- 26 M. Li, F. Igbari, Z. K. Wang and L. S. Liao, *Adv. Energy Mater.*, 2020, **10**, 2000641.
- 27 G. Létay and A. Bett, *Spectrum*, 2001, **20**, 25.
- 28 T. C. J. Yang, P. Fiala, Q. Jeangros and C. Ballif, *Joule*, 2018, **2**, 1421–1436.
- 29 J. Kim, J. H. Jang, E. Choi, S. J. Shin, J. H. Kim, G. G. Jeon and N. Park, *Cell Rep. Phys. Sci.*, 2020, **1**, 100273.
- 30 G. Lucarelli, F. Di Giacomo, V. Zardetto, M. Creatore and T. M. Brown, *Nano Res.*, 2017, **10**, 2130–2145.
- 31 K. Sharma, V. Sharma and S. S. Sharma, *Nanoscale Res. Lett.*, 2018, **13**, 1–46.
- 32 M. Freunek and L. M. Reindl, *IEEE J. Photovoltaics*, 2013, **3**, 1464.
- 33 I. Mathews, P. J. King, F. Stafford and R. Frizzell, *IEEE J. Photovoltaics*, 2015, **6**, 230–235.
- 34 H. S. Jung and N. G. Park, *Small*, 2015, **11**, 10–25.
- 35 L. Meng, J. You and Y. Yang, *Nat. Commun.*, 2018, **9**, 1–4.
- 36 C. Zhao, W. Tian, Q. Sun, Z. Yin, J. Leng, S. Wang and S. Jin, *J. Am. Chem. Soc.*, 2020, **142**, 15091–15097.
- 37 A. Karmakar, M. S. Dodd, S. Agnihotri, E. Ravera and V. K. Michaelis, *Chem. Mater.*, 2018, **30**, 8280–8290.
- 38 N. J. Jeon, J. H. Noh, W. S. Yang, Y. C. Kim, S. Ryu, J. Seo and S. I. Seok, *Nature*, 2015, **517**, 476–480.
- 39 F. C. Chen, *Adv. Opt. Mater.*, 2019, **7**, 1800662.
- 40 J. L. Wu, F. C. Chen, M. K. Chuang and K. S. Tan, *Energy Environ. Sci.*, 2011, **4**, 3374–3378.
- 41 G. Dennler, S. Bereznev, D. Fichou, K. Holl, D. Ilic, R. Koeppel and T. Wöhrle, *Sol. Energy*, 2007, **81**, 947–957.
- 42 J. F. Randall and J. Jacot, *Renewable Energy*, 2003, **28**, 1851–1864.
- 43 R. Santbergen, J. M. Goud, M. Zeman, J. A. M. van Roosmalen and R. J. C. van Zolingen, *Sol. Energy Mater. Sol. Cells*, 2010, **94**, 715–723.
- 44 X. Ma, S. Bader and B. Oelmann, *IEEE Sens. J.*, 2017, **17**, 3884–3891.
- 45 I. Mathews, S. N. Kantareddy, T. Buonassisi and I. M. Peters, *Joule*, 2019, **3**, 1415–1426.
- 46 G. Apostolou, *Procedia Environ. Sci.*, 2017, **38**, 905–912.
- 47 L. K. Ma, Y. Chen, P. C. Y. Chow, G. Zhang, J. Huang, C. Ma, J. Zhang, H. Yin, A. M. Hong Cheung, K. S. Wong, S. K. So and H. Yan, *Joule*, 2020, **4**, 1486–1500.
- 48 S. Castro-Hermosa, G. Lucarelli, M. Top, M. Fahland, J. Fahlteich and T. M. Brown, *Cell Rep. Phys. Sci.*, 2020, **1**, 100045.
- 49 V. C. Y. Chen, W. H. Lee, S. Y. Hsiao, W. L. Tsai, L. Yang, H. L. Lin, H. J. Chou and H. W. Lin, *J. Mater. Chem. A*, 2019, **7**, 3612–3617.
- 50 K. Kawata, K. Tamaki and M. Kawaraya, *J. Photopolym. Sci. Technol.*, 2015, **28**, 415–417.
- 51 M. Fath, A. Mefoued, A. Messaoud and Y. Boukennous, *Phys. Procedia*, 2009, **2**, 751.
- 52 Amorphous Solar Panel, [https://www.wsl-solar.com/Amorphous\\_Solar\\_Panel/](https://www.wsl-solar.com/Amorphous_Solar_Panel/), accessed April 2020.
- 53 Thin-Film Solar Cells for Low and High Illumination, <https://www.com/en/indoor-solar-cells>, accessed April 2020.
- 54 Indoor Solar Panels, <https://www.powerfilmsolar.com/specialty-markets/indoor-solar-panels/>, accessed April 2020.
- 55 Ricoh Launches the World's First Solid-State Dye-Sensitized Solar Cell Modules, [https://www.ricoh.com/release/2020/0204\\_1/](https://www.ricoh.com/release/2020/0204_1/), accessed April 2020.
- 56 Indoor Dye Sensitized Solar Cells, <https://gcell.com/dye-sensi-tized-solar-cells/advantages-of-dscc/indoor-dye-sensitize-solar-cells>, accessed April 2020.
- 57 BCC Research, *Wireless sensors: technologies and global markets*, BCC Research, 2016, <https://www.bccresearch.com/market-research/instrumentation-and-sensors/wireless-sensors-technologies-report-ias019c.html>.
- 58 P. D. Moskowitz and V. M. Fthenakis, *Sol. Cells*, 1990, **29**, 63–71.
- 59 A. Babayigit, A. Ethirajan, M. Muller and B. Conings, *Nat. Mater.*, 2016, **15**, 247–251.
- 60 A. Babayigit, D. Duy Thanh, A. Ethirajan, J. Manca, M. Muller, H. G. Boyen and B. Conings, *Sci. Rep.*, 2016, **6**, 1–11.
- 61 J. d. C. Silva, J. J. P. C. Rodrigues, A. M. Alberti, P. Solic and A. L. L. Aquino, *presented at 2nd Int. Multidisciplinary Conf. on Computer and Energy Science (SpliTech), XXXX, Split, Croatia, July 2017*.
- 62 BCC Research, *Global Markets, Technologies and Devices for Energy Harvesting: EGY097C*, <https://www.prnewswire.com/news-releases/global-markets-technologies-and-devices-for-energy-har-vesting-300677825.html>, accessed December 2019.
- 63 D. M. Powell, R. Fu, K. Horowitz, P. A. Basore, M. Woodhouse and T. Buonassisi, *Energy Environ. Sci.*, 2015, **8**, 3395–3408.
- 64 M. O. Reese, S. Glynn, M. D. Kempe, D. L. McGott, M. S. Dabney, T. M. Barnes, S. Booth, D. Feldman and N. M. Haegel, *Nat. Energy*, 2018, **3**, 1002–1012.
- 65 L. Lin, S. Jain and M. AlioTso, in *2018 IEEE International Solid-State Circuits Conference-ISSCC*, IEEE, 2018, pp. 44–46.
- 66 M. Schuß, C. A. Boano, M. Weber and K. Römer, in *EWSN*, 2017, pp. 54–65.
- 67 BCC Research, *Global Markets, Technologies and Devices for Energy Harvesting: EGY097C*, 2018, <https://www.bccresearch.com/market-research/energy-and-resources/global-markets-technologies-and-devices-for-energy-harvesting-egy097c.html>.
- 68 Technology Platforms for the Internet of Things (IoT), <https://www.bccresearch.com/market-research/information-technology/technology-platforms-for-the-internet-of-things-IoTs.html>, accessed April 2020.



- 69 W. Shockley and H. J. Queisser, *J. Appl. Phys.*, 1961, **32**, 510–519.
- 70 A. S. Teran, J. Wong, W. Lim, G. Kim, Y. Lee, D. Blaauw and J. D. Phillips, *IEEE Trans. Electron Devices*, 2015, **62**, 2170–2175.
- 71 Y. Cui, H. Yao, T. Zhang, L. Hong, B. Gao, K. Xian, J. Qin and J. Hou, *Adv. Mater.*, 2019, **31**, 1904512.
- 72 J. Peng, L. Lu, H. Yang and T. Ma, *Renewable Energy*, 2015, **80**, 316–323.
- 73 Y. Li, N. J. Grabham, S. P. Beeby and M. J. Tudor, *Sol. Energy*, 2015, **111**, 21–29.
- 74 N. H. Reich, W. G. J. H. M. Van Sark, E. A. Alsema, R. W. Lof, R. E. I. Schropp, W. C. Sinke and W. C. Turkenburg, *Sol. Energy Mater. Sol. Cells*, 2009, **93**, 1471–1481.
- 75 C. Y. Chen, Z. H. Jian, S. H. Huang, K. M. Lee, M. H. Kao, C. H. Shen, J. M. Shieh, C. L. Wang, C. W. Chang and B. Z. Lin, *J. Phys. Chem. Lett.*, 2017, **8**, 1824–1830.
- 76 Y. Hu, S. Si, A. Mei, Y. Rong, H. Liu, X. Li and H. Han, *Sol. RRL*, 2017, **1**, 1600019.
- 77 H. K. Lee, Z. Li, J. R. Durrant and W. C. Tsoi, *Appl. Phys. Lett.*, 2016, **108**, 253301.
- 78 K. Yoshikawa, H. Kawasaki, W. Yoshida, T. Irie, K. Konishi, K. Nakano and K. Yamamoto, *Nat. Energy*, 2017, **2**, 1–8.
- 79 M. Freunek, M. Freunek and L. M. Reindl, *IEEE J. Photovoltaics*, 2012, **3**, 59–64.
- 80 V. Bahrami-Yekta and T. Tiedje, *Opt. Express*, 2018, **26**, 28238–28248.
- 81 M. H. Kao, C. H. Shen, P. C. Yu, W. H. Huang, Y. L. Chueh and J. M. Shieh, *Sci. Rep.*, 2017, **7**, 1–8.
- 82 M. Kasemann, K. Rühle, K. M. Gad and S. W. Glunz, *Photovoltaic energy harvesting for smart sensor systems*, International Society for Optics and Photonics, 2013.
- 83 Q. Li, K. Shen, R. Yang, Y. Zhao, S. Lu, R. Wang, J. Dong and D. Wang, *Sol. Energy*, 2017, **157**, 216–226.
- 84 V. Ramanathan, L. A. Russell, C. H. Liu and P. V. Meyers, *Sol. Cells*, 1990, **28**, 129–133.
- 85 P. D. Antunez, D. M. Bishop, Y. Luo and R. Haight, *Nat. Energy*, 2017, **2**, 884–890.
- 86 C. Yang, J. Qu and Z. Wu, *Sol. Energy*, 2021, **214**, 542–550.
- 87 M. A. Green, Y. Hishikawa, E. D. Dunlop, D. H. Levi, J. Hohl-Ebinger and A. W. Y. Ho-Baillie, *Prog. Photovoltaics Res. Appl.*, 2018, **26**, 427–436.
- 88 I. Mathews, P. J. King, F. Stafford and R. Frizzell, *IEEE J. Photovoltaics*, 2016, **6**, 230–235.
- 89 I. Mathews, G. Kelly, P. J. King and R. Frizzell, *2014 IEEE 40th Photovoltaic Specialist Conference (PVSC)*, IEEE, 2014, vol. 40, pp. 0510–0513.
- 90 M. Zhang, L. Zhu, T. Hao, G. Zhou, C. Qiu, Z. Zhao and F. Liu, *Adv. Mater.*, 2021, 2007177.
- 91 H. K. H. Lee, J. Wu, J. Barbé, S. M. Jain, S. Wood, E. M. Speller, Z. Li, F. A. Castro, J. R. Durrant and W. C. Tsoi, *J. Mater. Chem. A*, 2018, **6**, 5618–5626.
- 92 Y. Cao, Y. Liu, S. M. Zakeeruddin, A. Hagfeldt and M. Grätzel, *Joule*, 2018, **2**, 1108–1117.
- 93 M. Freitag, J. Teuscher, Y. Saygili, X. Zhang, F. Giordano, P. Liska, J. Hua, S. M. Zakeeruddin, J. E. Moser and M. Grätzel, *Nat. Photonics*, 2017, **11**, 372–378.
- 94 H. Michaels, M. Rinderle, R. Freitag, I. Benesperi, T. Edvinsson, R. Socher and M. Freitag, *Chem. Sci.*, 2020, **11**, 2895–2906.
- 95 D. Zhang, M. Stojanovic, Y. Ren, Y. Cao, F. T. Eickemeyer, E. Socie and M. Grätzel, *Nat. Commun.*, 2021, **12**, 1–10.
- 96 Z. Ding, R. Zhao, Y. Yu and J. Liu, *J. Mater. Chem. A*, 2019, **7**, 26533–26539.
- 97 S. N. Agbo, T. Merdzhanova, U. Rau and O. Astakhov, *Sol. Energy Mater. Sol. Cells*, 2017, **159**, 427–434.
- 98 A. S. Teran, E. Moon, W. Lim, G. Kim, I. Lee, D. Blaauw and J. D. Phillips, *IEEE Trans. Electron Devices*, 2016, **63**, 2820–2825.
- 99 Y. Dai, H. Kum, M. A. Slocum, G. T. Nelson and S. M. Hubbard, *presented in part at the IEEE 44th Photovoltaic Specialist Conference (PVSC) 2017*.
- 100 M. H. Ann, J. Kim, M. Kim, G. Alosaimi, D. Kim, N. Y. Ha and J. H. Kim, *Nano Energy*, 2020, **68**, 104321.
- 101 Y. Cao, Y. Liu, S. M. Zakeeruddin, A. Hagfeldt and M. Grätzel, *Joule*, 2018, **2**, 1108–1117.
- 102 H. Yin, S. Chen, S. H. Cheung, H. W. Li, Y. Xie, S. W. Tsang and S. K. So, *J. Mater. Chem. C*, 2018, **6**, 9111–9118.
- 103 L. J. A. Koster, V. D. Mihailetschi, H. Xie and P. W. M. Blom, *Appl. Phys. Lett.*, 2005, **87**, 203502.
- 104 Z. Li, G. Lakhwani, N. C. Greenham and C. R. McNeill, *J. Appl. Phys.*, 2013, **114**, 034502.
- 105 A. K. K. Kyaw, D. H. Wang, V. Gupta, W. L. Leong, L. Ke, G. C. Bazan and A. J. Heeger, *ACS Nano*, 2013, **7**, 4569.
- 106 L. J. A. Koster, V. D. Mihailetschi, R. Ramaker and P. W. M. Blom, *Appl. Phys. Lett.*, 2005, **86**, 123509.
- 107 S. R. Cowan, A. Roy and A. J. Heeger, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2010, **82**, 245207.
- 108 T. Kirchartz, F. Deledalle, P. S. Tuladhar, J. R. Durrant and J. Nelson, *J. Phys. Chem. Lett.*, 2013, **4**, 2371–2376.
- 109 Y. J. You, C. E. Song, Q. V. Hoang, Y. Kang, J. S. Goo, D. H. Ko, J. J. Lee, W. S. Shin and J. W. Shim, *Adv. Funct. Mater.*, 2019, **29**, 1901171.
- 110 Y. Cui, Y. Wang, J. Bergqvist, H. Yao, Y. Xu, B. Gao, C. Yang, S. Zhang, O. Inganäs, F. Gao and J. Hou, *Nat. Energy*, 2019, **4**, 768–775.
- 111 R. Steim, T. Ameri, P. Schilinsky, C. Waldauf, G. Dennler, M. Scharber and C. J. Brabec, *Sol. Energy Mater. Sol. Cells*, 2011, **95**, 3256.
- 112 S. Mori, T. Gotanda, Y. Nakano, M. Saito, K. Todori and M. Hosoya, *Jpn. J. Appl. Phys.*, 2015, **54**, 071602.
- 113 A. Venkateswararao, J. K. W. Ho, S. K. So, S. W. Liu and K. T. Wong, *Mater. Sci. Eng., R*, 2020, **139**, 100517.
- 114 Y. Cui, L. Hong, T. Zhang, H. Meng, H. Yan, F. Gao and J. Hou, *Joule*, 2021, **5**, 1016–1023.
- 115 Z. Song, C. L. McElvany, A. B. Phillips, I. Celik, P. W. Krantz, S. C. Wathage and M. J. Heben, *Energy Environ. Sci.*, 2017, **10**, 1297–1305.
- 116 O. Haillant, *Sol. Energy Mater. Sol. Cells*, 2011, **95**, 1284.
- 117 G. Grancini, C. Roldán-Carmona, I. Zimmermann, E. Mosconi, X. Lee, D. Martineau and M. K. Nazeeruddin, *Nat. Commun.*, 2017, **8**, 1–8.
- 118 X. He, J. Chen, X. Ren, L. Zhang, Y. Liu, J. Feng and S. Liu, *Adv. Mater.*, 2021, 2100770.



- 119 S. A. Gevorgyan, M. V. Madsen, B. Roth, M. Corazza, M. Hösel, R. R. Søndergaard and F. C. Krebs, *Adv. Energy Mater.*, 2016, **6**, 1501208.
- 120 S. S. Shin, E. J. Yeom, W. S. Yang, S. Hur, M. G. Kim, J. Im and S. I. Seok, *Science*, 2017, **356**, 167–171.
- 121 W. Li, W. Zhang, S. Van Reenen, R. J. Sutton, J. Fan, A. A. Haghighirad and H. J. Snaith, *Energy Environ. Sci.*, 2016, **9**, 490–498.
- 122 S. Ito, S. Tanaka, K. Manabe and H. Nishino, *J. Phys. Chem. C*, 2014, **118**, 16995–17000.
- 123 C. J. Mulligan, M. Wilson, G. Bryant, B. Vaughan, X. Zhou, W. J. Belcher and P. C. Dastoor, *Sol. Energy Mater. Sol. Cells*, 2014, **120**, 9.
- 124 C. J. Mulligan, M. Wilson, G. Bryant, B. Vaughan, X. Zhou, W. J. Belcher and P. C. Dastoor, *Sol. Energy Mater. Sol. Cells*, 2014, **120**, 9–17.
- 125 C. Mayousse, C. Celle, A. Fraczkiewicz and J.-P. Simonato, *Nanoscale*, 2015, **7**, 2107–2115.
- 126 D. S. Hecht, L. Hu and G. Irvin, *Adv. Mater.*, 2011, **23**, 1482–1513.
- 127 B. T. Camic, H. J. Shin, M. H. Aslan, F. Basarir and H. Choi, *J. Colloid Interface Sci.*, 2018, **512**, 158–164.
- 128 S. Kim, H. Oh, G. Kang, I. K. Han, I. Jeong and M. Park, *ACS Appl. Energy Mater.*, 2020, **3**, 6995–7003.
- 129 J. Dagar, S. Castro-Hermosa, M. Gasbarri, A. L. Palma, L. Cina, F. Matteocci and T. M. Brown, *Nano Res.*, 2018, **11**, 2669–2681.
- 130 F. Qin, J. Tong, R. Ge, B. Luo, F. Jiang, T. Liu and Y. Zhou, *J. Mater. Chem. A*, 2016, **4**, 14017–14024.
- 131 K. K. Sears, M. Fievez, M. Gao, H. C. Weerasinghe, C. D. Easton and D. Vak, *Sol. RRL*, 2017, **1**, 1700059.
- 132 X. L. Ou, M. Xu, J. Feng and H. B. Sun, *Sol. Energy Mater. Sol. Cells*, 2016, **157**, 660–665.
- 133 S. P. Cho, S. L. Na and S. S. Kim, *Sol. Energy Mater. Sol. Cells*, 2019, **196**, 1–8.
- 134 T. Aernouts, P. Vanlaeke, W. Geens, J. Poortmans, P. Heremans, S. Borghs, R. Mertens, R. Andriessen and L. Leenders, *Thin Solid Films*, 2004, **22**, 451–452.
- 135 G. Dennler, C. Lungenschmied, H. Neugebauer, N. S. Sariciftci and A. Labouret, *J. Mater. Res.*, 2005, **20**, 3224.
- 136 N. G. Park and K. Zhu, *Nat. Rev. Mater.*, 2020, **5**, 333–350.
- 137 NREL Chart, in, June 2019.
- 138 H. J. Snaith, *J. Phys. Chem. Lett.*, 2013, **4**, 3623–3630.
- 139 B. A. Nejjand, P. Nazari, S. Gharibzadeh, V. Ahmadi and A. Moshaii, *Chem. Commun.*, 2017, **53**, 747–750.
- 140 H. Tang, S. He and C. Peng, *Nanoscale Res. Lett.*, 2017, **12**, 1–8.
- 141 L. Cai, L. Liang, J. Wu, B. Ding, L. Gao and B. Fan, *J. Semicond.*, 2020, **38**, 014006.
- 142 S. Razza, S. Castro-Hermosa, A. Di Carlo and T. M. Brown, *APL Mater.*, 2016, **4**, 091508.
- 143 J. Burschka, N. Pellet, S. J. Moon, R. Humphry-Baker, P. Gao, M. K. Nazeeruddin and M. Grätzel, *Nature*, 2013, **499**, 316–319.
- 144 F. De Rossi, T. Pontecorvo and T. M. Brown, *Appl. Energy*, 2015, **156**, 413–422.
- 145 K. Sun, J. Chang, F. H. Isikgor, P. Li and J. Ouyang, *Nanoscale*, 2015, **7**, 896–900.
- 146 H. D. Pham, S. M. Jain, M. Li, Z. K. Wang, S. Manzhos, K. Feron and P. Sonar, *Adv. Electron. Mater.*, 2020, **6**, 1900884.
- 147 Y. W. Noh, I. S. Jin, K. S. Kim, S. H. Park and J. W. Jung, *J. Mater. Chem. A*, 2020, **8**, 17163–17173.
- 148 R. Cheacharoen, C. C. Boyd, G. F. Burkhard, T. Leijtens, J. A. Raiford, K. A. Bush and M. D. McGehee, *Sustainable Energy Fuels*, 2018, **2**, 2398–2406.
- 149 H. C. Weerasinghe, Y. Dkhissi, A. D. Scully, R. A. Caruso and Y. B. Cheng, *Nano Energy*, 2015, **18**, 118–125.
- 150 A. Uddin, M. B. Upama, H. Yi and L. Duan, *Coatings*, 2019, **9**, 65.
- 151 C. B. Wang, F. D. Gu, Z. R. Zhao, H. X. Rao, Y. M. Qiu, Z. L. Cai, G. Zhan, X. Y. Li, B. X. Sun, X. Yu, B. Q. Zhao, Z. W. Liu, Z. Q. Bian and C. H. Huang, *Adv. Mater.*, 2020, **32**, 1907623.
- 152 J. P. Cao and Y. Feng, *Energy Environ. Sci.*, 2021, **14**, 1286–1325.
- 153 H. Yao, F. Zhou, Z. Li, Z. Ci, L. Ding and Z. Jin, *Adv. Sci.*, 2020, **7**, 1903540.
- 154 B. Chen, X. Zheng, Y. Bai, N. P. Padture and J. Huang, *Adv. Energy Mater.*, 2017, **7**, 1602400.
- 155 M. Anaya, G. Lozano, M. E. Calvo and H. Míguez, *Joule*, 2017, **1**, 769–793.
- 156 T. C.-J. Yang, P. Fiala, Q. Jeangros and C. Ballif, *Joule*, 2018, **2**, 1421–1436.
- 157 N. N. Lal, Y. Dkhissi, W. Li, Q. Hou, Y.-B. Cheng and U. Bach, *Adv. Energy Mater.*, 2017, **7**, 1602761.
- 158 E. L. Unger, L. Kegelmann, K. Suchan, D. Sörell, L. Korte and S. Albrecht, *J. Mater. Chem. A*, 2017, **5**, 11401–11409.
- 159 Z. Wang, Z. Song, Y. Yan, S. Liu and D. Yang, *Adv. Sci.*, 2019, **6**, 1801704.
- 160 Z. Song, C. Chen, C. Li, R. A. Awni, D. Zhao and Y. Yan, *Semicond. Sci. Technol.*, 2019, **34**, 093001.
- 161 A. Kojima, K. Teshima, Y. Shirai and T. Miyasaka, *J. Am. Chem. Soc.*, 2009, **131**, 6050–6051.
- 162 M. M. Lee, J. Teuscher, T. Miyasaka, T. N. Murakami and H. J. Snaith, *Science*, 2012, **338**, 643–647.
- 163 G. Kieslich, S. Sun and A. K. Cheetham, *Chem. Sci.*, 2015, **6**, 3430–3433.
- 164 M. Saliba, J. P. Correa-Baena, M. Grätzel, A. Hagfeldt and A. Abate, *Angew. Chem., Int. Ed.*, 2018, **57**, 2554–2569.
- 165 <https://www.nrel.gov/pv/assets/pdfs/best-research-cell-efficiencies.20200218.pdf>.
- 166 M. A. Green, *Nat. Energy*, 2016, **1**, 15015.
- 167 C. C. Chueh, C. Z. Li and A. K. Y. Jen, *Energy Environ. Sci.*, 2015, **8**, 1160–1189.
- 168 Y. Deng, X. Zheng, Y. Bai, Q. Wang, J. Zhao and J. Huang, *Nat. Energy*, 2018, **3**, 560–566.
- 169 J. P. Correa-Baena, A. Abate, M. Saliba, W. Tress, T. J. Jacobsson, M. Grätzel and A. Hagfeldt, *Energy Environ. Sci.*, 2017, **10**, 710–727.
- 170 J. Gong, S. B. Darling and F. You, *Energy Environ. Sci.*, 2015, **8**, 1953–1968.



- 171 N. Espinosa, L. Serrano-Luján, A. Urbina and F. C. Krebs, *Sol. Energy Mater. Sol. Cells*, 2015, **137**, 303–310.
- 172 I. Celik, Z. Song, A. J. Cimaroli, Y. Yan, M. J. Heben and D. Apul, *Sol. Energy Mater. Sol. Cells*, 2016, **156**, 157–169.
- 173 K. P. Bhandari, J. M. Collier, R. J. Ellingson and D. S. Apul, *Renewable Sustainable Energy Rev.*, 2015, **47**, 133–141.
- 174 M. Cai, Y. Wu, H. Chen, X. Yang, Y. Qiang and L. Han, *Adv. Sci.*, 2017, **4**, 1600269.
- 175 N. L. Chang, A. W. Yi Ho-Baillie, P. A. Basore, T. L. Young, R. Evans and R. J. Egan, *Prog. Photovoltaics*, 2017, **25**, 390–405.
- 176 Y. Chen, L. Zhang, Y. Zhang, H. Gao and H. Yan, *RSC Adv.*, 2018, **8**, 10489–10508.
- 177 Q. Jiang, Z. Chu, P. Wang, X. Yang, H. Liu, Y. Wang, Z. Yin, J. Wu, X. Zhang and J. You, *Adv. Mater.*, 2017, **29**, 1703852.
- 178 K. Hwang, Y. S. Jung, Y. J. Heo, F. H. Scholes, S. E. Watkins, J. Subbiah, D. J. Jones, D. Y. Kim and D. Vak, *Adv. Mater.*, 2015, **27**, 1241–1247.
- 179 J. G. Tait, S. Manghooli, W. Qiu, L. Rakocevic, L. Kootstra, M. Jaysankar, C. A. Massede la Huerta, U. W. Paetzold, R. Gehlhaar, D. Cheyins, P. Heremans and J. Poortmans, *J. Mater. Chem. A*, 2016, **4**, 3792–3797.
- 180 O. A. Basaran, H. Gao and P. P. Bhat, *Annu. Rev. Fluid Mech.*, 2013, **45**, 85–113.
- 181 F. D. Giacomo, S. Shanmugam, H. Fledderus, B. J. Bruijnaers, W. J. Verhees, M. S. Dorenkamper, S. C. Veenstra, W. Qiu, R. Gehlhaar and T. Merckx, *Sol. Energy Mater. Sol. Cells*, 2018, **181**, 53–59.
- 182 Z. Liu, L. Qiu, E. J. J. Perez, Z. Hawash, T. Kim, Y. Jiang, Z. Wu, S. R. Raga, L. K. Ono, S. F. Liu and Y. Qi, *Nat. Commun.*, 2018, **9**, 3880.
- 183 C. Zuo, D. Vak, D. Angmo, L. Ding and M. Gao, *Nano Energy*, 2018, **46**, 185–192.
- 184 J. E. Kim, S. S. Kim, C. Zuo, M. Gao, D. Vak and D. Y. Kim, *Adv. Funct. Mater.*, 2019, **29**, 1809194.
- 185 Y. Y. Kim, T. Y. Yang, R. Suhonen, M. Välimäki, T. Maaninen, A. Kemppainen, N. J. Jeon and J. Seo, *Adv. Sci.*, 2019, **6**, 1802094.
- 186 W. W. Liu, T. H. Wu, M. C. Liu, W. J. Niu and Y. L. Chueh, *Adv. Mater. Interfaces*, 2019, **6**, 1801758.
- 187 Z. Cheng and J. Lin, *CrystEngComm*, 2010, **12**, 2646.
- 188 I. C. Smith, E. T. Hoke, D. Solis-Ibarra, M. D. McGehee and H. I. Karunadasa, *Angew. Chem., Int. Ed.*, 2014, **53**, 1.
- 189 L. J. Zuo, Q. Chen, N. D. Marco, Y. T. Hsieh, H. J. Chen, P. Y. Sun, S. Y. Chang, H. X. Zhao, S. Q. Dong and Y. Yang, *Nano Lett.*, 2017, **17**, 269.
- 190 T. Miyasaka, A. Kulkarni, G. M. Kim, S. Öz and A. K. Jena, *Adv. Energy Mater.*, 2020, **10**, 1902500.
- 191 A. Abate, *Joule*, 2017, **1**, 659–664.
- 192 A. Singh, P. T. Lai, A. Mohapatra, C. Y. Chen, H. W. Lin, Y. J. Lu and C. W. Chu, *Chem. Eng. J.*, 2021, **419**, 129424.
- 193 G. Kapil, T. Bessho, T. Maekawa, A. K. Baranwal, Y. Zhang, M. A. Kamarudin and S. Hayase, *Adv. Energy Mater.*, 2021, 2101069.
- 194 C. Wang, Y. Zhang, F. Gu, Z. Zhao, H. Li, H. Jiang and Z. Liu, *Matter*, 2021, **4**, 709–721.
- 195 W. F. Yang, J. J. Cao, C. Dong, M. Li, Q. S. Tian, Z. K. Wang and L. S. Liao, *Appl. Phys. Lett.*, 2021, **118**, 023501.
- 196 S. Y. Park, J. S. Park, B. J. Kim, H. Lee, A. Walsh, K. Zhu and H. S. Jung, *Nature Sustainability*, 2020, **3**, 1044–1051.
- 197 F. Terroso-Saenz, A. González-Vidal, A. P. Ramallo-González and A. F. Skarmeta, *Future Gener. Comput. Syst.*, 2019, **92**, 1066–1079.
- 198 D. L. DiLaura, K. Houser, R. Mistrick and G. R. Steffy, *The lighting handbook: reference and application*, 2011.
- 199 B. Minnaert and P. Veelaert, *Energies*, 2014, **7**, 1500.
- 200 W. J. Yin, T. Shi and Y. Yan, *Appl. Phys. Lett.*, 2014, **104**, 063903.
- 201 C. Dong, M. Li, Y. Zhang, K. L. Wang, S. Yuan, F. Igbari, Y. G. Yang, X. Y. Gao, Z. K. Wang and L. S. Liao, *ACS Appl. Mater. Interfaces*, 2019, **12**, 836–843.
- 202 R. Cheng, C. C. Chung, H. Zhang, F. Liu, W. T. Wang, Z. Zhou, S. Wang, A. B. Djurišić and S. P. Feng, *Adv. Energy Mater.*, 2019, **9**, 1901980.
- 203 M. J. Wu, C. C. Kuo, L. S. Jhuang, P. H. Chen, Y. F. Lai and F. C. Chen, *Adv. Energy Mater.*, 2019, **9**, 1901863.
- 204 J. W. Lim, H. Kwon, S. H. Kim, Y. J. You, J. S. Goo, D. H. Ko, H. J. Lee, D. Kim, I. Chung, T. G. Kim, D. H. Kim and J. W. Shim, *Nano Energy*, 2020, **75**, 104984.
- 205 Y. Peng, T. N. Huq, J. Mei, L. Portilla, R. A. Jagt, L. G. Occhipinti, J. L. MacManus-Driscoll, R. L. Z. Hoye and V. Pecunia, *Adv. Energy Mater.*, 2021, **11**, 2002761.
- 206 E. Jokar, C. H. Chien, C. M. Tsai, A. Fathi and E. W. G. Diau, *Adv. Mater.*, 2019, **31**, 1804835.
- 207 L. Serrano-Lujan, N. Espinosa, T. T. Larsen-Olsen, J. Abad, A. Urbina and F. C. Krebs, *Adv. Energy Mater.*, 2015, **5**, 1501119.
- 208 Q. Hu, J. Wu, C. Jiang, T. Liu, X. Que, R. Zhu and Q. Gong, *ACS Nano*, 2014, **8**, 10161.
- 209 P. Schulz, E. Edri, S. Kirmayer, G. Hodes, D. Cahen and A. Kahn, *Energy Environ. Sci.*, 2014, **7**, 1377.
- 210 G. Yang, H. Tao, P. Qin, W. Ke and G. Fang, *J. Mater. Chem. A*, 2016, **4**, 3970–3990.
- 211 J. Shi, X. Xu, D. Li and Q. Meng, *Small*, 2015, **11**, 2472.
- 212 C. C. Boyd, R. Cheacharoen, T. Leijtens and M. D. McGehee, *Chem. Rev.*, 2018, **119**, 3418.
- 213 M. Li, C. Zhao, Z. K. Wang, C. C. Zhang, H. K. H. Lee, A. Pockett, J. Barbé, W. C. Tsoi, Y. G. Yang, M. J. Carnie, X. Y. Gao, W. X. Yang, J. R. Durrant, L. S. Liao and S. M. Jain, *Adv. Energy Mater.*, 2018, **8**, 1801509.
- 214 C. Dong, X. M. Li, C. Ma, W. F. Yang, J. J. Cao, F. Igbari, Z. K. Wang and L. S. Liao, *Adv. Funct. Mater.*, 2021, **31**, 2011242.
- 215 K. L. Wang, X. M. Li, Y. H. Lou, M. Li and Z. K. Wang, *Sci. Bull.*, 2021, **66**, 347–353.
- 216 Z. Li, J. Zhang, S. F. Wu, X. Deng, F. Z. Li, D. J. Liu, C. C. Lee, F. Lin, D. Y. Lei, C. C. Chueh, Z. L. Zhu and A. K. Y. Jen, *Nano Energy*, 2020, **78**, 105377.
- 217 H. K. H. Lee, J. Barbé, S. M. Meroni, T. Du, C. T. Lin, A. Pockett, J. Troughton, S. M. Jain, F. D. Rossi, J. Baker, M. J. Carnie, M. A. McLachlan, T. M. Watson, J. R. Durrant and W. C. Tsoi, *Sol. RRL*, 2019, **3**, 1800207.



- 218 L. K. Jagadamma, O. Blaszczyk, M. T. Sajjad, A. Ruseckas and I. D. Samuel, *Sol. Energy Mater. Sol. Cells*, 2019, **201**, 110071.
- 219 W. Ke, G. Fang, Q. Liu, L. Xiong, P. Qin, H. Tao, J. Wang, H. Lei, B. Li, J. Wan, G. Yang and Y. Yan, *J. Am. Chem. Soc.*, 2015, **137**, 6730–6733.
- 220 J. Burschka, N. Pellet, S. J. Moon, R. Humphry-Baker, P. Gao, M. K. Nazeeruddin and M. Grätzel, *Nature*, 2013, **499**, 316–319.
- 221 S. Razza, F. Di Giacomo, F. Matteocci, L. Cinà, A. L. Palma, S. Casaluci, P. Cameron, A. D'Epifanio, S. Licoccia, A. Reale, T. M. Brown and A. Di Carlo, *J. Power Sources*, 2015, **277**, 286–291.
- 222 B. Conings, L. Baeten, T. Jacobs, R. Dera, J. D'Haen, J. Manca and H. G. Boyen, *APL Mater.*, 2014, **2**, 081505.
- 223 F. Di Giacomo, V. Zardetto, G. Lucarelli, L. Cinà, A. Di Carlo, M. Creatore and T. M. Brown, *Nano Energy*, 2016, **30**, 460–469.
- 224 P. Tiwana, P. Docampo, M. B. Johnston, H. J. Snaith and L. M. Herz, *ACS Nano*, 2011, **5**, 5158–5166.
- 225 H. J. Snaith and C. Ducati, *Nano Lett.*, 2010, **10**, 1259–1265.
- 226 F. Di Giacomo, A. Fakharuddin, R. Jose and T. M. Brown, *Energy Environ. Sci.*, 2016, **9**, 3007–3035.
- 227 J. Dagar, S. Castro-Hermosa, G. Lucarelli, F. Cacialli and T. M. Brown, *Nano Energy*, 2018, **49**, 290–299.

