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## Exploring the thermoelectric properties of oligo(phenylene-ethynylene) derivatives†

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Seebeck coefficient measurements provide unique insights into the electronic structure of single-molecule junctions, which underpins their charge and heat transport properties. Since the Seebeck coefficient depends on the slope of the transmission function at the Fermi energy ( $E_F$ ), the sign of the thermoelectric voltage will be determined by the location of the molecular orbital levels relative to  $E_F$ . Here we investigate thermoelectricity in molecular junctions formed from a series of oligophenylene-ethynylene (OPE) derivatives with biphenylene, naphthalene and anthracene cores and pyridyl or methylthio end-groups. Single-molecule conductance and thermoelectric voltage data were obtained using a home-built scanning tunneling microscope break junction technique. The results show that all the OPE derivatives studied here are dominated by the lowest unoccupied molecular orbital level. The Seebeck coefficients for these molecules follow the same trend as the energy derivatives of their corresponding transmission spectra around the Fermi level. The molecule terminated with pyridyl units has the largest Seebeck coefficient corresponding to the highest slope of the transmission function at  $E_F$ . Density-functional-theory-based quantum transport calculations support the experimental results.

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## Introduction

Thermoelectric materials enable direct conversion between thermal energy and electrical energy and open a new way to harvest new environmentally-friendly energy for energy-savings and recycling. Nanoscale organic molecular thermoelectric materials are promising due to their tunable electronic structure and the feasibility of large-scale production through chemical synthesis.<sup>1,2</sup> Thus, studies of charge and heat transport in single-molecule junctions, formed by trapping an indi-

vidual molecule between two electrodes, are of great fundamental importance to developing molecular electronics and energy-conversion devices.<sup>3–9</sup> Intensive experimental and theoretical studies have tried to reveal the fundamental mechanisms of charge and thermal transport at the nanoscale, and recent progress has suggested that organic materials at the single-molecule scale can show enhanced thermoelectric properties by adjusting their charge transport properties.<sup>10–13</sup> The measurement of the Seebeck coefficient allows us to determine the ability of heat-to-electricity conversion at the organic junction. Furthermore, theory indicates that electrical transport through molecular junctions can be described by the Landauer formula<sup>14,15</sup> and the Seebeck coefficient is given in the scattering formalism by<sup>16,17</sup>

$$S(E_F) = -\frac{\pi^2 k_B^2 T}{3|e|} \left. \frac{\partial T(E)}{\partial E} \right|_{E=E_F} \quad (1)$$

where  $T(E)$  is the energy-dependent transmission function,  $T$  is the average temperature of the leads,  $E_F$  is the Fermi level of the leads and  $k_B$  is the Boltzmann constant. The Seebeck coefficient is proportional to the energy derivative of the transmission coefficient in eqn (1).<sup>16</sup> This fact indicates that the Seebeck coefficient can also be used to determine the dominant channel of a molecular junction. The sign of the coeffi-

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cient will establish which molecular energy level is closer to  $E_F$ , either the highest occupied molecular orbital (HOMO) (if the measured value is positive) or the lowest unoccupied molecular orbital (LUMO) (if it is negative).<sup>15</sup> Therefore, the single-molecule electronics techniques offer a unique opportunity to investigate the thermoelectric properties of organic materials, which allows further promotion of organic energy conversion devices.

Since the thermoelectric properties of a molecular junction were investigated through the STM break-junction (STM-BJ),<sup>18</sup> scanning thermal microscopy (SThM),<sup>19</sup> mechanically controlled break junction (MCBJ)<sup>20</sup> and electromigration<sup>21</sup> techniques, those advances have transformed studies of thermal and thermoelectric properties at the single-molecule scale and even enabled fabrication of single-molecule energy conversion devices, thermal rectifiers or transistors.<sup>19,22–27</sup> The thermoelectric properties of molecules can be tuned by molecular length,<sup>28–30</sup> electrode coupling,<sup>31–33</sup> quantum interference effects,<sup>34–36</sup> intermolecular interactions,<sup>37</sup> substituent effects,<sup>38</sup> variations of the end group,<sup>24</sup> and also by using pi-stacked systems to increase the heat current, controlling the molecular energy levels relative to the Fermi energy of the electrodes.<sup>12</sup> Further investigation of the thermoelectric properties of a family of molecules with varying electronic structures will provide the experimental and theoretical insights into the correlation between the transmission function and the Seebeck coefficient based on eqn (1).

Here we investigate the Seebeck coefficient of four oligo (phenylene-ethynylene) (OPE) derivatives shown in Fig. 1a using a home-built scanning tunnelling microscope break

junction technique (STM-BJ).<sup>18</sup> All the molecules have 1,4-difunctionalized pyridyl<sup>39</sup> or methylthio<sup>28</sup> terminals at both ends and the OPE cores in their backbone.<sup>40</sup> The choice of these four molecules is motivated by their expected different energy derivatives in electron-transmission characteristics. Seebeck coefficient measurements show that the transport through all the four molecules is dominated by their LUMO channels. The Seebeck coefficients for these molecules follow the same trend as the energy derivatives of their corresponding transmission spectra around the Fermi level and are supported by our Seebeck coefficient calculations.

## Experimental section

The synthesis of molecules 1–4 has been reported previously.<sup>40</sup> Their single-molecule charge transport properties have been determined at room temperature using a home-built scanning tunnelling microscope break junction technique (STM-BJ).<sup>18</sup> The target molecule was self-assembled onto a 30 mm × 10 mm silicon substrate (with a 200 nm thick gold film on it), a fixed bias voltage of 100 mV was applied between the gold tip and the substrate (see ESI† for details). ~2000 individual conductance–distance traces without any selection were recorded for each molecule for further statistical analysis. A blank control experiment was performed using a clean gold substrate without any molecule assembled, which suggested that the background noise was around  $10^{-6}G_0$ . In order to prevent the contact between the tip and the substrate to avoid direct heat transfer during the thermoelectric experiment, a

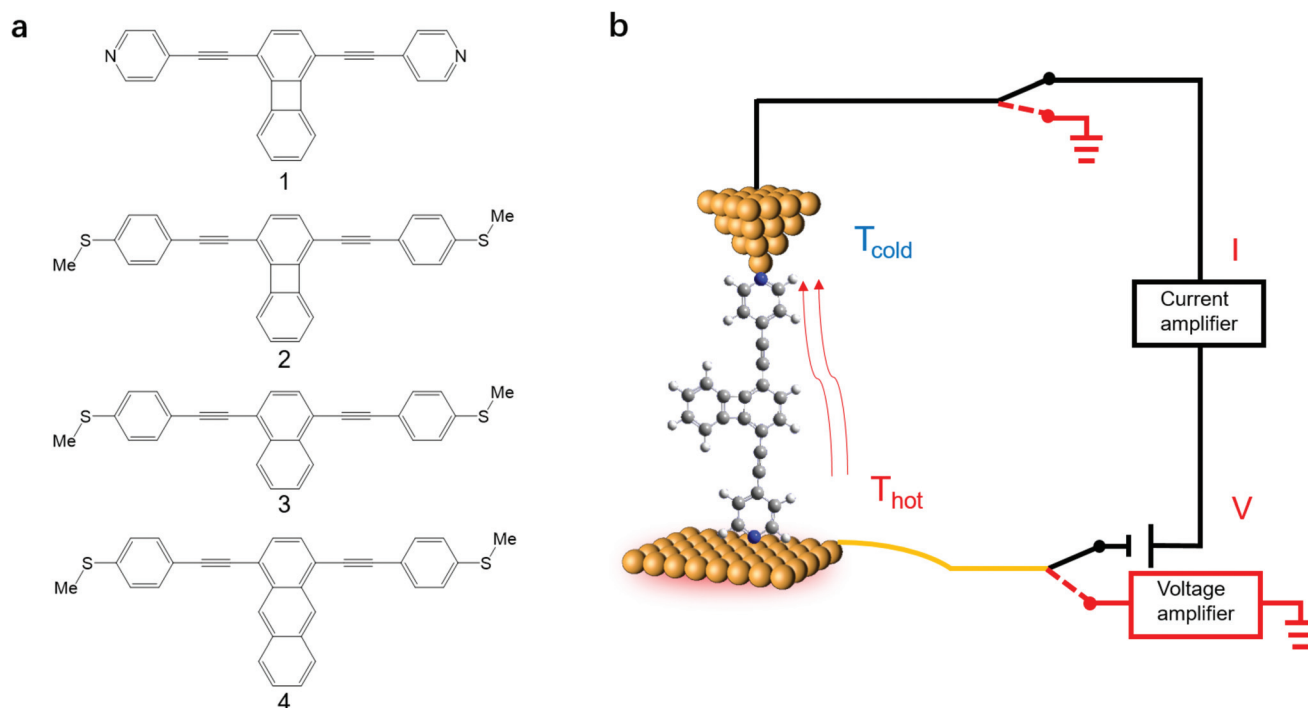


Fig. 1 (a) The structures of OPE derivatives 1–4 studied in this work. (b) Schematic of the experimental setup.



soft contact mode was chosen to measure the molecular conductance and thermopower (see ESI† for details).<sup>41,42</sup> A high-threshold conductance value of  $0.01G_0$  (where  $G_0$  is the quantum conductance, which equals  $2e^2/h$ ) was chosen as the high trigger to control the Au tip approaching and withdrawing, so that only single-molecule junctions are formed.

To investigate the Seebeck coefficient of single-molecule devices, we incorporated a home-built temperature control unit into the STM-BJ setup.<sup>23,34</sup> The scheme of the home-built STM-BJ set-up is shown in Fig. 1b, and the set-up incorporated a conductance mode (conductance mode: only current amplifier switched on) and a thermoelectricity mode (thermoelectricity mode: only voltage amplifier switched on). To create a stable temperature difference between the tip and the substrate, the substrate was heated *via* a Peltier device, and the temperature difference was modulated by a thermocouple mounted under the substrate with feedback control. Meanwhile, the tip was kept at room temperature.<sup>23,43</sup> A series of temperature differences ( $\Delta T = \Delta T_{\text{substrate}} - \Delta T_{\text{tip}} \approx 0, 5, 10$  to 15 K) between the tip and the substrate were set respectively in the air. During the thermoelectric measurements, the tip, controlled by a piezoelectric actuator, was withdrawn from the substrate until a stable molecular junction was formed. Once the plateau was created, the circuit was switched from conductance mode to thermoelectricity mode through a relay switch. The thermoelectric voltage ( $\Delta V_{\text{th}} = \Delta V_{\text{substrate}} - \Delta V_{\text{tip}}$ ) across a single-molecule junction was directly recorded using a voltage amplifier for a period of 100 ms, which was induced by the corresponding temperature difference. Then, the molecular junction broke down, and the set-up switched back to conductance mode. Histograms of the thermoelectric voltage were used to estimate the variation of the Seebeck coefficient and then normalized to make sure each temperature difference contains the same number of counts. Gaussian fitting gave the most probable thermoelectric voltage  $\Delta V_{\text{th}}$ . The heated substrate was connected to the voltage amplifier using copper wires, which contribute an additional thermoelectric voltage to

the system. Therefore, the Seebeck coefficient of the molecular junction function is given by<sup>34</sup>

$$S_{\text{junction}} = S_{\text{Cu}} - \frac{\Delta V_{\text{th}}}{\Delta T} \quad (2)$$

where  $S_{\text{Cu}} = 1.94 \mu\text{V K}^{-1}$  is the Seebeck coefficient of bulk copper at  $T = 300 \text{ K}$ .<sup>44</sup> (See ESI† for all details.)

## Results and discussion

Fig. 2a shows the typical individual traces of conductance ( $G$ ) *versus* stretching distance ( $\Delta z$ ) plotted in a semi-logarithmic scale. The single-molecule conductance plateaus of molecules 1–4 using the soft-contact mode vary from  $10^{-4.73}$  to  $10^{-4.30}G_0$ . It can be clearly seen from Table 1 that these values are very similar to the results reported previously<sup>40</sup> using a previous MCBJ method in their corresponding molecular solutions with hard contact of electrodes. Similar molecules with alkynyl spacers within the backbone exhibit similar conductance data in MCBJ or STM-BJ experiments in previous work.<sup>39,45</sup> This is because the electrode coupling has little influence on the molecular backbone. We thus confirmed that the present method of soft contact to measure the assembled molecules in the air was reliable. To quantitatively determine the most probable conductance values, Fig. 2b indicates the one-dimensional (1D) conductance histograms of the four molecules, which reveal that the most probable conductance values for molecules 1–4 were  $10^{-4.73}G_0$ ,  $10^{-4.52}G_0$ ,  $10^{-4.32}G_0$ , and  $10^{-4.30}G_0$ , respectively. The plateaus and peaks were further confirmed by a clear density cloud in the corresponding 2D conductance–distance histograms, as shown in Fig. 2c for compound 1. It can be clearly seen that the molecule 2 with methylthio anchoring groups shows a higher conductance value than pyridyl-terminated molecule 1, as observed previously.<sup>40</sup> The similar conductance values of molecules 2 to 4 reveal that the pendant cyclobutadiene unit of 2 has no signifi-



Fig. 2 Single-molecule conductance results from soft-contact STM-BJ experiments. (a) Typical measured individual conductance–displacement traces from soft-contact break junction measurements for compounds 1–4 and the blank pure-solvent control experiment (magenta), respectively. (b) 1D conductance histograms constructed from 2000 conductance–distance traces without any selection. (c) 2D conductance histograms of compound 1.



**Table 1** Single-molecule Seebeck coefficient and conductance measurements

Compounds	Seebeck coefficient <sup>a</sup> ( $\mu\text{V K}^{-1}$ )	Measured conductance/ $\log(G/G_0)^b$ -STM-BJ	Measured conductance/ $\log(G/G_0)^b$ -MCBJ <sup>40</sup>
1	$-9.71 \pm 1.06$	$-4.73 \pm 0.22$	$-4.6 \pm 0.41$
2	$-6.88 \pm 0.67$	$-4.52 \pm 0.23$	$-4.4 \pm 0.52$
3	$-1.62 \pm 0.21$	$-4.32 \pm 0.18$	$-4.2 \pm 0.46$
4	$-4.13 \pm 1.52$	$-4.30 \pm 0.33$	$-4.1 \pm 0.49$

<sup>a</sup>The error bars are based on the standard deviation in the linear fitting of the most probable thermoelectric voltage as a function of  $\Delta T$ .

<sup>b</sup>Most probable conductance values and the error bars are based on the standard deviation in the Gaussian fitting of the 1D conductance histograms.

cant impact on charge transport. Moreover, in contrast to our previous MCBJ results,<sup>40</sup> we observed only a high conductance peak of molecule 4, suggesting that the signal is from single molecules.

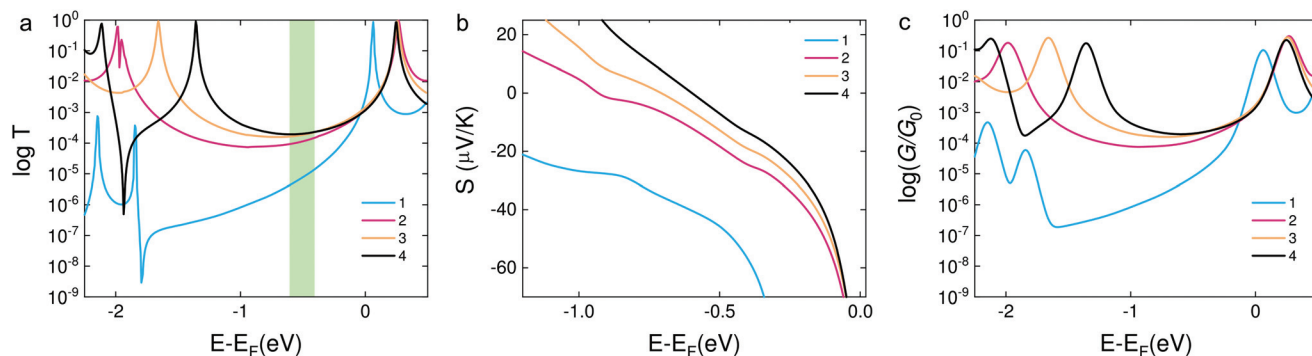
Typical histograms of the thermoelectric voltage for molecule 1 are shown in Fig. 3a. Since the fluctuation of molecular junction configurations is inevitable during the measurement process in ambient environment, the fluctuation of the molecular configuration and the electrode contact coupling is likely

to lead to the fluctuation of the Seebeck coefficient.<sup>46,47</sup> Therefore, the thermoelectric voltage from different molecular junctions might exhibit different distributions. Thus the thermoelectric voltage values of more than 1000 junctions were recorded and summarized into histograms during the thermoelectric voltage measurement. The histogram peaks, which are based on the Gaussian fitting, representing the most probable measured  $V_{\text{peak}}$  were chosen and plotted as a function of  $\Delta T$ , and the Seebeck coefficient was obtained from the thermoelectric slope as shown in Fig. 3b. These values are plotted in Fig. 3c and summarized in Table 1. The Seebeck coefficients for molecules 1–4 are  $-9.71 \pm 1.06 \mu\text{V K}^{-1}$ ,  $-6.88 \pm 0.67 \mu\text{V K}^{-1}$ ,  $-1.62 \pm 0.21 \mu\text{V K}^{-1}$  and  $-4.13 \pm 1.52 \mu\text{V K}^{-1}$ , respectively (see ESI, Fig. S5† for more details). It can be clearly seen that the sign of the Seebeck coefficients for all four OPE derivatives is negative, which means that  $E_{\text{F}}$  is closer to the lowest unoccupied molecular orbital (LUMO) level and their transport properties are electron-dominated. The Seebeck coefficient value for molecule 1 was found to be the largest, followed by molecules  $2 > 4 > 3$ . Moreover, the methylthio-terminal was found to impart HOMO-dominated transport<sup>28</sup> or LUMO-dominated transport<sup>48</sup> in the previous studies, whereas in these molecular systems, molecules 1–4 all show LUMO-dominated transport (negative Seebeck coefficient). The only difference between 1



**Fig. 3** (a) Histograms of thermoelectric voltage measurements for compound 1 at a series of  $\Delta T = 0$  K, 5 K, 10 K and 15 K, respectively. Gaussian fits plotted as a black short dash line. Blue dash line indicates the Gaussian fitting centre of thermoelectric voltage at  $\Delta T = 0$  K. (b) The Seebeck coefficients obtained from the thermoelectric voltage as a function of  $\Delta T$ . Red solid lines are the linear fitting. The error bars are based on the standard deviation in the Gaussian fitting of thermoelectric voltage. (c) Plot of all conductance (black circles) and the Seebeck coefficients (red squares) with error bars for molecules 1–4.





**Fig. 4** (a) DFT results of the transmission coefficients for compounds 1–4. (b) Seebeck coefficients of the corresponding molecules. The highlighted area shows the Fermi energy at which the calculated transmission and Seebeck coefficient are in qualitative agreement with the experimental findings. The relaxed structures are shown in Fig. S8 in the ESI.† (c) Calculated electrical conductance of molecules 1–4.

and 2 is the anchor group, but the magnitude of the Seebeck coefficient of molecule 1 is much higher (more negative) than molecule 2. Considering the above-mentioned factor, we suppose such a difference in the thermoelectric performance of the two molecules is caused by destructive quantum interference in molecule 1.

In order to confirm the hypothesis that the destructive quantum interference occurred in molecule 1, we used density functional theory (DFT) to obtain the mean field Hamiltonian from the relaxed ground state geometry of the junctions (see Methods in the ESI†) and calculated the transmission coefficient  $T(E)$  of electrons with energy  $E$  passing from one electrode to the other. From the calculated  $T(E)$  value shown in Fig. 4a, an antiresonance peak is observed between the HOMO and LUMO of molecule 1, which causes a steeper slope around the Fermi level of the electrode. However, the transmission functions of molecule 2 show a typical feature of constructive quantum interference, suggesting that the significant difference between molecules 1 and 2 is due to the quantum interference effect.

Calculations show that the conductance of  $4 \approx 3 > 2$  is in good agreement with the experiment for a wide energy range around  $E - E_F = -0.5$  eV. From the transmission functions in Fig. 4a, we can attribute the trend to the reduction of the HOMO–LUMO gap. The LUMO is closer to the Fermi level, suggesting that the transport channels of all three molecules are dominated by the LUMO which is consistent with the negative Seebeck coefficient measured in our experiments. The LUMOs of molecules 2–4 show no obvious changes with the different pendant units on the central phenylene ring. However, the HOMO of the three molecules is sequentially shifted towards the Fermi level, and this phenomenon was also observed in the investigation of a series of thiophene-1,1-dioxide derivatives.<sup>28</sup> Thus, the conductance increased and the Seebeck coefficient decreased caused by the LUMO moving closer to the Fermi level.

We used the calculated  $T(E)$  value to obtain the Seebeck coefficients shown in Fig. 4b and the corresponding electrical conductance shown in Fig. 4c. We considered that all the

Fermi energies of the electrodes in the case of each molecule are in the same range, as indicated by a green band in Fig. 4a. This yields the correct trend for molecules 1, 2 and 4. Molecule 3 is an outlier and has a smaller Seebeck coefficient than expected, which indicates that the Fermi energy in this case is closer to the middle of the HOMO–LUMO gap. However, the predicted values of the Seebeck coefficient were higher than the measured values. In order to correct the level alignment obtained from DFT, a scissor correction was performed using the method described in ref. 49 and 50. This leads to a better agreement between the theory and the experiment as shown in Fig. S8 in the ESI.† We found that the Seebeck coefficient of 1 is about two times higher than that of 2 for a wide energy range. Furthermore, the Seebeck coefficient of 4 is higher than that of 3; both of the above results are in agreement with the experiment. The higher overall values obtained from calculations when compared to the experimental results can be induced by different binding configurations to the electrodes in the experiment which lead to energy level broadening. The values of the Seebeck coefficient are sensitive to the slope of  $T(E)$  as shown in eqn (1), and therefore are more sensitive to the broadening of resonances compared to the electrical conductance.

## Conclusions

In conclusion, we studied the thermoelectric properties of OPE derivatives using a modified STM-BJ technique, and we demonstrated the importance of the anchor groups in the OPE derivatives. We further established that the increase in pendant unit conjugation of OPE derivatives would lead to a decrease in the Seebeck coefficient. The *ab initio* calculations are in good agreement with the experimental results. Furthermore, since the three-terminal configurations with a gating voltage can actively tune the transmission behaviours of molecular junctions, this should be a good strategy for the design of future organic molecular devices with efficient thermoelectric energy conversion.



## Author contributions

W. H., M. R. B. and C. J. L. conceived the concept and co-supervised the project. STM-BJ setup was constructed by H. C. and G. L. in W. H.'s group. H. C. and G. L. carried out single-molecule conductance and thermoelectric experiments. S.S., A. A. and H. S. performed theoretical simulations. M. G. and G. O. completed synthesis of molecules used in this work. C. Z. and Y. Z. helped to fabricate the substrate. W. C., J. L., L. C. and Y. C. helped to analyse data. H. C. and S. S. wrote the manuscript. H. C., S. S., J. L., G. O., M. R. B., C. J. L. and W. H. helped to revise and polish the manuscript. All authors gave approval to the final version of the manuscript.

## Conflicts of interest

The authors declare no competing financial interest.

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## Notes and references

- B. Russ, A. Glauddell, J. J. Urban, M. L. Chabinye and R. A. Segalman, *Nat. Rev. Mater.*, 2016, **1**, 16050.
- Y.-J. Zeng, D. Wu, X.-H. Cao, W.-X. Zhou, L.-M. Tang and K.-Q. Chen, *Adv. Funct. Mater.*, 2020, **30**, 1903873.
- A. R. Nitzan and M. A. Ratner, *Science*, 2003, **300**, 1384–1389.
- D. Xiang, X. Wang, C. Jia, T. Lee and X. Guo, *Chem. Rev.*, 2016, **116**, 4318–4440.
- S. V. Aradhya and L. Venkataraman, *Nat. Nanotechnol.*, 2013, **8**, 399–410.
- E. Leary, A. La Rosa, M. T. González, G. Rubio-Bollinger, N. Agraït and N. Martín, *Chem. Soc. Rev.*, 2015, **44**, 920–942.
- R. J. Nichols and S. J. Higgins, *Annu. Rev. Anal. Chem.*, 2015, **8**, 389–417.
- M. A. Reed, C. Zhou, C. J. Muller, T. P. Burgin and J. M. Tour, *Science*, 1997, **278**, 252–254.
- F. Chen, J. Hihath, Z. Huang, X. Li and N. J. Tao, *Annu. Rev. Phys. Chem.*, 2007, **58**, 535–564.
- C. M. Finch, V. M. García-Suárez and C. J. Lambert, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2009, **79**, 033405.
- Q. H. Al-Galiby, H. Sadeghi, L. A. Algharagholy, I. Grace and C. Lambert, *Nanoscale*, 2016, **8**, 2428–2433.
- G. Kiršanskas, Q. Li, K. Flensberg, G. C. Solomon and M. Leijnse, *Appl. Phys. Lett.*, 2014, **105**, 233102.
- Q. H. Al-Galiby, H. Sadeghi, D. Z. Manrique and C. J. Lambert, *Nanoscale*, 2017, **9**, 4819–4825.
- M. Buttiker, Y. Imry, R. Landauer and S. Pinhas, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1985, **31**, 6207–6215.
- M. Paulsson and S. Datta, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2003, **67**, 241403.
- U. Sivan and Y. Imry, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 1986, **33**, 551–558.
- P. N. Butcher, *J. Phys. Chem.*, 1990, **2**, 4869–4878.
- B. Xu and N. J. Tao, *Science*, 2003, **301**, 1221–1223.
- L. Cui, W. Jeong, S. Hur, M. Matt, J. C. Klockner, F. Pauly, P. Nielaba, J. C. Cuevas, E. Meyhofer and P. Reddy, *Science*, 2017, **355**, 1192–1195.
- T. Morikawa, A. Arima, M. Tsutsui and M. Taniguchi, *Nanoscale*, 2014, **6**, 8235–8241.
- Y. Kim, W. Jeong, K. Kim, W. Lee and P. Reddy, *Nat. Nanotechnol.*, 2014, **9**, 881–885.
- L. Cui, R. Miao, K. Wang, D. Thompson, L. A. Zotti, J. C. Cuevas, E. Meyhofer and P. Reddy, *Nat. Nanotechnol.*, 2017, **13**, 122–127.
- P. Reddy, S.-Y. Jang, R. A. Segalman and A. Majumdar, *Science*, 2007, **315**, 1568–1571.
- J. R. Widawsky, P. Darancet, J. B. Neaton and L. Venkataraman, *Nano Lett.*, 2012, **12**, 354–358.
- L. Cui, S. Hur, Z. A. Akbar, J. C. Klockner, W. Jeong, F. Pauly, S. Y. Jang, P. Reddy and E. Meyhofer, *Nature*, 2019, **572**, 628–633.
- N. Mosso, U. Drechsler, F. Menges, P. Nirmalraj, S. Karg, H. Riel and B. Gotsmann, *Nat. Nanotechnol.*, 2017, **12**, 430–433.
- N. Mosso, H. Sadeghi, A. Gemma, S. Sangtarash, U. Drechsler, C. Lambert and B. Gotsmann, *Nano Lett.*, 2019, **19**, 7614–7622.
- E. J. Dell, B. Capozzi, J. Xia, L. Venkataraman and L. M. Campos, *Nat. Chem.*, 2015, **7**, 209–214.
- T. Meier, F. Menges, P. Nirmalraj, H. Holscher, H. Riel and B. Gotsmann, *Phys. Rev. Lett.*, 2014, **113**, 060801.
- J. R. Widawsky, W. Chen, H. Vazquez, T. Kim, R. Breslow, M. S. Hybertsen and L. Venkataraman, *Nano Lett.*, 2013, **13**, 2889–2894.
- S. K. Yee, J. A. Malen, A. Majumdar and R. A. Segalman, *Nano Lett.*, 2011, **11**, 4089–4094.
- R.-N. Wang, G.-Y. Dong, S.-F. Wang, G.-S. Fu and J.-L. Wang, *J. Appl. Phys.*, 2016, **120**, 184303.
- Y.-X. Zhen, M. Yang and R.-N. Wang, *Front. Phys.*, 2019, **14**, 23603.



- 34 R. Miao, H. Xu, M. Skripnik, L. Cui, K. Wang, K. G. L. Pedersen, M. Leijnse, F. Pauly, K. Warnmark, E. Meyhofer, P. Reddy and H. Linke, *Nano Lett.*, 2018, **18**, 5666–5672.
- 35 C. J. Lambert, H. Sadeghi and Q. H. Al-Galiby, *C. R. Phys.*, 2016, **17**, 1084–1095.
- 36 C. J. Lambert, *Chem. Soc. Rev.*, 2015, **44**, 875–888.
- 37 C. Evangeli, K. Gillemot, E. Leary, M. T. Gonzalez, G. Rubio-Bollinger, C. J. Lambert and N. Agrait, *Nano Lett.*, 2013, **13**, 2141–2145.
- 38 K. Baheti, J. A. Malen, P. Doak, P. Reddy, S. Y. Jang, T. D. Tilley, A. Majumdar and R. A. Segalman, *Nano Lett.*, 2008, **8**, 715–719.
- 39 W. Hong, D. Z. Manrique, P. Moreno-Garcia, M. Gulcur, A. Mishchenko, C. J. Lambert, M. R. Bryce and T. Wandlowski, *J. Am. Chem. Soc.*, 2012, **134**, 2292–2304.
- 40 M. Gantenbein, X. Li, S. Sangtarash, J. Bai, G. Olsen, A. Alqorashi, W. Hong, C. J. Lambert and M. R. Bryce, *Nanoscale*, 2019, **11**, 20659–20666.
- 41 P. Moreno-García, A. La Rosa, V. Kolivoška, D. Bermejo, W. Hong, K. Yoshida, M. Baghernejad, S. Filippone, P. Broekmann, T. Wandlowski and N. Martín, *J. Am. Chem. Soc.*, 2015, **137**, 2318–2327.
- 42 W. Haiss, H. van Zalinge, S. J. Higgins, D. Bethell, H. Höbenreich, D. J. Schiffrin and R. J. Nichols, *J. Am. Chem. Soc.*, 2003, **125**, 15294–15295.
- 43 L. Shi and A. Majumdar, *J. Heat Transfer*, 2002, **124**, 329–337.
- 44 F. J. Blatt, *Thermoelectric Power of Metals*, Plenum Press, New York, 1976.
- 45 P. Duan, J. Liu, J.-Y. Wang, K. Qu, S. Cai, F. Wang, L. Chen, X. Huang, R. Li, J. Shi, Q.-C. Zhang, W. Hong and Z.-N. Chen, *Sci. China: Chem.*, 2020, **63**, 467–474.
- 46 L. Rincon-Garcia, A. K. Ismael, C. Evangeli, I. Grace, G. Rubio-Bollinger, K. Porfyrikis, N. Agrait and C. J. Lambert, *Nat. Mater.*, 2016, **15**, 289–293.
- 47 R.-N. Wang, G.-Y. Dong, S.-F. Wang, G.-S. Fu and J.-L. Wang, *Phys. Chem. Chem. Phys.*, 2016, **18**, 28117–28124.
- 48 R. J. Davidson, D. C. Milan, O. A. Al-Owaedi, A. K. Ismael, R. J. Nichols, S. J. Higgins, C. J. Lambert, D. S. Yufit and A. Beeby, *RSC Adv.*, 2018, **8**, 23585–23590.
- 49 H. Sadeghi, *Nanotechnology*, 2018, **29**, 373001.
- 50 J. Ferrer, C. J. Lambert, V. M. García-Suárez, D. Z. Manrique, D. Visontai, L. Oroszlany, R. Rodríguez-Ferradás, I. Grace, S. W. D. Bailey, K. Gillemot, H. Sadeghi and L. A. Algharagholy, *New J. Phys.*, 2014, **16**, 093029.

