

Cite this: *Chem. Sci.*, 2022, 13, 4955

All publication charges for this article have been paid for by the Royal Society of Chemistry

Photoelectrochemical water oxidation improved by pyridine *N*-oxide as a mimic of tyrosine-Z in photosystem II†‡

Yong Zhu,^a Guoquan Liu,^a Ran Zhao,^a Hua Gao,^a Xiaona Li,^b Licheng Sun^{b,acd} and Fei Li^{b,*a}

Artificial photosynthesis provides a way to store solar energy in chemical bonds with water oxidation as a major challenge for creating highly efficient and robust photoanodes that mimic photosystem II. We report here an easily available pyridine *N*-oxide (PNO) derivative as an efficient electron transfer relay between an organic light absorber and molecular water oxidation catalyst on a nanoparticle TiO₂ photoanode. Spectroscopic and kinetic studies revealed that the PNO/PNO^{•+} couple closely mimics the redox behavior of the tyrosine/tyrosyl radical pair in PSII in improving light-driven charge separation via multi-step electron transfer. The integrated photoanode exhibited a 1 sun current density of 3 mA cm⁻² in the presence of Na₂SO₃ and a highly stable photocurrent density of >0.5 mA cm⁻² at 0.4 V vs. NHE over a period of 1 h for water oxidation at pH 7. The performance shown here is superior to those of previously reported organic dye-based photoanodes in terms of photocurrent and stability.

Received 24th January 2022

Accepted 31st March 2022

DOI: 10.1039/d2sc00443g

rsc.li/chemical-science

Introduction

Photocatalytic splitting of water into hydrogen and oxygen is one of the most attractive routes for solar energy conversion. Among the available methodologies, the photoelectrochemical (PEC) cell based on a semiconductor/liquid junction has been recognized as a promising approach enabling the production of solar fuels at relatively low cost.^{1–3} In PEC cells, the oxidation of water is a major obstacle of overall water splitting due to the demand for multi-electron and multi-proton transfer. Although a large body of PEC cells has been developed based on light-harvesting inorganic semiconductors, a highly efficient photoanode is yet to be explored. As an alternative approach, dye-sensitized photoelectrochemical cells (DSPECs) have received wide attention considering their ability to be engineered at the molecular level.⁴ On a TiO₂-based DSPEC photoanode, the surface-bound visible dye is integrated with a water oxidation

catalyst (WOC). Photoexcitation of the dye leads to electron injection into the conduction band of the TiO₂ substrate, followed by cross-surface electron migration from the WOC to the oxidized dye. The latter step is particularly challenging because of the need to compete with back-electron transfer at the oxide interface.^{5–7}

In the past decade, the phosphonate-derivatized tris(2,2'-bipyridine)ruthenium(II) dichloride salt (RuP) has been applied as a typical photosensitizer for PEC water oxidation.⁸ Despite favorable photophysical properties, the employment of a noble metal hindered its practical application in artificial photosynthesis. Aiming at scalable application and economic viability, a DSPEC photoanode should be cost efficient to produce O₂ gas. In this regard, organic dyes are attractive candidates in terms of their low cost, earth-abundance and structural tunability by well-established organic synthesis techniques. Previously, a variety of organic dyes including perylene diimide,^{9,10} porphyrins,^{11–16} Janus green B¹⁷ and triphenylamine-based donor-acceptor compounds^{18–21} have been integrated in DSPECs for visible-light harvesting. However, organic dyes often suffer from a short excited-state lifetime, and their performance towards solar-driven water oxidation remains poor. The reported photocurrent densities achieved by organic dye-based photoanodes fall in a range of 20 to 200 μA cm⁻², the magnitudes of which are far less than that achieved by RuP (up to 1.7 mA cm⁻² was reported).²² In addition, compared with noble metal-based photosensitizers, organic dyes are more vulnerable to oxidative degradation, leading to low stability in aqueous environments. In order to overcome these limitations, metal oxides^{18,23} and polymers such as Nafion²⁴ and poly(methyl

^aState Key Laboratory of Fine Chemicals, Dalian University of Technology, Dalian 116024, China. E-mail: lifei@dlut.edu.cn

^bKey Laboratory of Industrial Ecology and Environmental Engineering (MOE), School of Environmental Science and Technology, Dalian University of Technology, Dalian 116024, China

^cDepartment of Chemistry, School of Engineering Sciences in Chemistry, Biotechnology and Health, KTH Royal Institute of Technology, Stockholm 10044, Sweden

^dCenter of Artificial Photosynthesis for Solar Fuels, School of Science, Westlake University, Hangzhou 310024, China

† This paper is dedicated to Prof. Licheng Sun on the occasion of his 60th birthday.

‡ Electronic supplementary information (ESI) available. See DOI: 10.1039/d2sc00443g



methacrylate) (PMMA)^{25,26} have been deposited on the surface of the TiO₂ electrode to stabilize organic dyes and related assemblies. However, the addition of protection layers cannot fully inhibit the decomposition of dyes upon light irradiation. Therefore, a more efficient method to improve the efficiency and stability of organic dyes for photoanodic water oxidation is urgently needed.

In the photosynthetic reaction center of photosystem II (PSII), tyrosine-Z (TyrZ) has been well recognized as a crucial cofactor to shuttle electrons between the CaMn₄O₄ oxygen evolution complex (OEC) and chromophore P680⁺.^{27–29} The tyrosine/tyrosyl radical redox couple mediated electron transfer results in quick recovery of the reduced state of the chromophore and thus efficiently protects the chromophore against decomposition.^{30,31} Another benefit of the redox mediator is that it efficiently retards unproductive back electron transfer by extending the distance between the chromophore and catalyst. In the configurations of DSPECs, synthetic models for the biological electron transfer relay have rarely been investigated. To the best of our knowledge, only two samples have been reported to date, and both are related to RuP sensitized-photoanodes. In one study, a benzimidazole-phenol moiety was covalently linked with surface-bound IrO₂ to accelerate light-induced electron transfer between the IrO₂ WOC and RuP.³² Recently, we

deposited an ultrathin NiO overlayer on a RuP sensitized SnO₂/TiO₂ film by atomic layer deposition (ALD), and the NiO layer served as an electron transfer relay between RuP and the exterior Ru-bda (H₂bda = 2,2'-bipyridine-6,6'-dicarboxylic acid) catalyst.³³ Despite these efforts, preparation of suitable redox mediators that efficiently reproduce the role of tyrosine in PSII remains a challenge.

We report herein that the performance of the organic dye and molecular WOC co-adsorbed TiO₂ photoanode could be considerably improved by incorporation with an easily available pyridine *N*-oxide (PNO) derivative as a molecular electron transfer analog for TyrZ (Fig. 1). During the catalytic cycle, the photogenerated pyridine *N*-oxide cation radical closely mimics the tyrosyl radical which efficiently extracts electrons from the WOC. With this design, the integrated TiO₂|dye|PNO|Cat film (Fig. 2a) exhibited superior photoanodic performance over other organic dye-based photoanodes in terms of photocurrent density and stability.

Results and discussion

During the fabrication of the photoanode, a donor–acceptor type triphenylamine (TPA) dye was chosen as the chromophore due to its sufficient driving force for water oxidation. The preparation and characterization of 18 nm nanoparticle TiO₂ films on FTO followed a literature procedure.²² As shown in Fig. 2b, the TPA dye was decorated with C5 alkyl chains and a phosphonate anchoring group, which was subsequently sensitized on a metal oxide substrate by immersing the TiO₂ thin film in a dye solution for 2 h. Based on the Lambert–Beer law, the coverage of dye on the TiO₂ film was 8×10^{-8} mol cm⁻², consistent with monolayer adsorption on the electrode surface.³⁴

The obtained electrode TiO₂/TPA was further decorated with 4-undecyl pyridine *N*-oxide (PNO) as an electron transfer relay (Fig. 2). Since the DSPEC was operated in aqueous solution, the modification of long alkyl chains on PNO allowed this mediator complex to be loaded as an overlayer *via* simple drop-casting,³⁵ a method that brings considerable convenience in fabrication

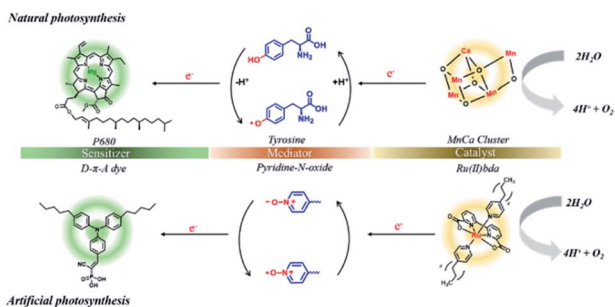


Fig. 1 Illustration of the role of the redox mediator in natural (top) and artificial (bottom) photosynthetic systems.



Fig. 2 (a) Schematic of the TiO₂|dye|PNO|Cat photoanode employed for water oxidation. (b) Molecular structures of TPA dye, PNO and Ru-based molecular water oxidation catalysts 1, 2 and 3 used in the study.



of molecular assemblies on metal oxides over the traditional ones *via* covalent links³² or ALD deposition.³³

The SEM image of TiO₂|TPA|PNO reveals a rough surface composed of amorphous nanoparticles and the cross-sectional SEM image of TiO₂|TPA|PNO shows a clear interface between mesoporous TiO₂ and the PNO layer, with an average thickness of 5 μm for the overlayer (Fig. 3). In comparison, a notable difference was observed for the polymer coated TiO₂ films. The SEM images of TiO₂|TPA|PMMA and TiO₂|TPA|Nafion show much flatter and denser membranes over the TiO₂ underlayer (Fig. S1†). Considering that the diameter of a TPA molecule is about 15 Å, the dyes were proposed to be completely buried in the PNO layer.

It has been reported that pyridine *N*-oxide can be electrochemically or photochemically oxidized in either aqueous or non-aqueous solutions, and the generation of a pyridine *N*-oxide radical cation was supported by EPR and other techniques.^{36,37} Agreeing with these findings, cyclic voltammetry (CV) of PNO modified nanoITO in neutral phosphate buffer shows an irreversible anodic wave at 1.1 V *vs.* NHE (Fig. S2†), which was ascribed to the oxidation of PNO to the cation radical PNO⁺. The effect of PNO on surface charge transfer was first investigated in the dark with TiO₂|TPA coated with different overlayers. The dynamic changes of the absorption spectra for each photoanode were recorded by imposing a constant bias of 1.2 V *vs.* NHE.^{18,38} During this process, intense absorption at 700 nm was observed for TiO₂|TPA, TiO₂|TPA|PMMA and TiO₂-|TPA|Nafion films due to the oxidation of TPA to TPA⁺ with

holes located on the dye (Fig. 4). By contrast, no TPA⁺ radical signal was observed for the electrode TiO₂|TPA|PNO during electrolysis. The spectroelectrochemical results obtained here suggest rapid dye regeneration by electron-donating PNO (eqn (1)), which forms the basis for light-driven charge separation at the hybrid interface.



The photoelectrochemical performance of TiO₂|TPA|PNO and its analogues TiO₂|TPA, TiO₂|TPA|PMMA and TiO₂-|TPA|Nafion was evaluated in a pH 7 phosphate buffer solution with 0.1 M Na₂SO₃ as a hole scavenger. The PEC experiments were conducted in a standard three electrode cell under 1 sun illumination (100 mW cm⁻²). Meanwhile, a 400 nm cutoff filter was used to avoid direct bandgap excitation of the underlying oxide. In the cell, a platinum mesh electrode was used as the counter electrode with an Ag/AgCl reference electrode. In Fig. 5a, the current densities of the four samples under dark/light cycling conditions follow the trend of TiO₂|TPA|PNO > TiO₂|TPA > TiO₂|TPA|PMMA ≈ TiO₂|TPA|Nafion. Impressively, a photocurrent of 3 mA cm⁻² was achieved by TiO₂|TPA|PNO at 0.4 V *vs.* NHE, which is 2–3 times that for PMMA and Nafion coated films. Since the oxidation of SO₃²⁻ is kinetically more facile than the oxidation of water, the density of photocurrent reflects the hole transport efficiency of a photoanode. Combined with the electrochemical results, this observation indicates that PNO possesses remarkable hole extraction ability, which is lacking for the polymers (PMMA and Nafion). Another notable feature of comparison is that rapid photocurrent decay occurs for the TiO₂|TPA film without a coating layer (Fig. 5b), while the other three samples exhibit stable photocurrents during the test, indicative of simultaneous improvement in hole migration and operation stability by adding PNO.

In order to realize water splitting, a Ru-bda based water oxidation catalyst (WOC) was chosen to improve the kinetics of oxygen evolution on a photoanode in view of its low overpotential and high activity under neutral conditions. To facilitate catalyst loading on the photoanode, a C11 alkyl chain was

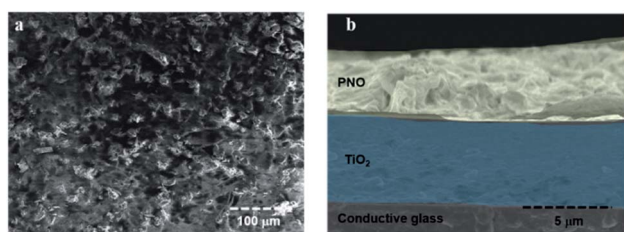


Fig. 3 SEM images of the TiO₂|TPA|PNO film, top view (a) and side view (b).

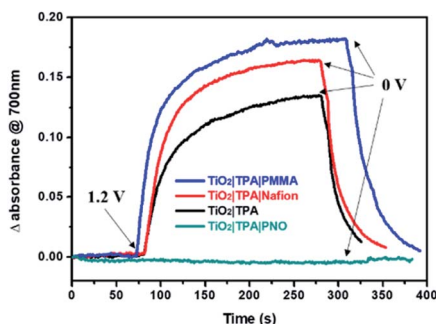


Fig. 4 Absorption changes at 700 nm associated with time-dependent oxidation of TiO₂|TPA|PMMA (blue), TiO₂|TPA|Nafion (red), TiO₂|TPA (black) and TiO₂|TPA|PNO (green) in pH 7, 0.1 M phosphate buffer containing 0.4 M NaClO₄ following a potential step from 1.2 to 0 V *vs.* NHE.

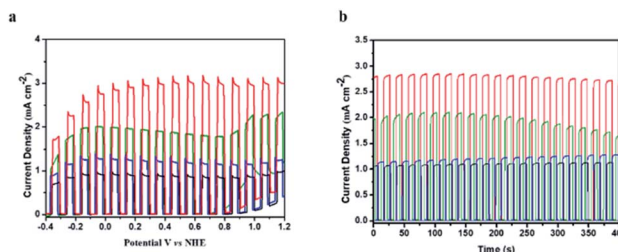


Fig. 5 (a) Photocurrent density–voltage (*j*–*v*) traces with 5 s light transients for TiO₂|TPA|PNO (red), TiO₂|TPA (green), TiO₂|TPA|Nafion (blue) and TiO₂|TPA|PMMA (black) scanned from –0.4 to 1.2 V *vs.* NHE with a scan rate of 1 mV s⁻¹ under 1 sun illumination in pH 7, 0.1 M phosphate buffer containing 0.1 M Na₂SO₃. (b) The time dependent photocurrent density (*j*–*t*) traces with 10 s light transients at 0.4 V *vs.* NHE.



modified on the axial pyridyl ligand of Ru-bda (**1**) (Fig. 2). Catalyst **1** was drop-cast on TiO₂|TPA|PNO at a surface loading of $\Gamma = 5 \times 10^{-8}$ mol cm⁻², a value comparable to that of previously reported surface-bound molecular catalysts on mesoporous TiO₂ films.²² The SEM image confirmed that the morphology of the PNO underlayer was fully maintained after catalyst loading (Fig. S3†).

The photoelectrochemical performance for water oxidation was evaluated under ambient conditions in phosphate buffer (pH 7). Linear sweep voltammetry (LSV) of TiO₂|TPA|PNO|**1** under 1 sun illumination resulted in a prominently enhanced photocurrent with a small onset potential at -0.18 V vs. NHE (Fig. 6a). As a typical behavior for a DSPEC, a saturated current above 500 $\mu\text{A cm}^{-2}$ is reached after passing a potential of 0.1 V vs. NHE. The effect of a PNO interlayer on photocurrent is notable. For example, TiO₂|TPA|**1** fabricated by co-loading of dye and catalyst **1** halves the saturated photocurrent (250 $\mu\text{A cm}^{-2}$). In our control experiment, TiO₂|TPA produced negligible photocurrent due to the lack of a catalyst.

The current density–time traces of these photoanodes under chopped illumination and a constant bias of 0.4 V vs. NHE are shown in Fig. 6b. The results again confirmed that the catalyst and light are indispensable factors for water oxidation. The comparison between TiO₂|TPA|PNO|**1** and TiO₂|TPA|**1** indicates a two-fold increase in the amplitude of the photocurrent. Another point worthy of note is that the photocurrent exhibited by TiO₂|TPA|PNO|**1** is more stable than that of TiO₂|TPA|**1**, consistent with the results of the sacrificial experiments. The improved durability as a consequence of insertion of an NPO interlayer between the catalyst and dye was further evidenced in an extended period of photolysis. For TiO₂|TPA|PNO|**1**, a stable

anodic photocurrent at 0.4 V vs. NHE, over 500 $\mu\text{A cm}^{-2}$, was maintained for at least 1 h without obvious decay (<10%), which is in sharp contrast to TiO₂|TPA|**1**, whose photocurrent density was significantly reduced by 70% during the same period due to the degradation of TPA by water molecules and hydroxide ions (Fig. 6c). To ascertain that the observed photocurrents are indeed produced by water splitting, a faradaic efficiency of 82% for O₂ production was calculated for long-term photolysis. The loss in faradaic efficiency is consistent with the decomposition and desorption of the catalyst which occur in parallel with water oxidation catalysis.²² The PEC experiment of TiO₂|TPA|PNO|**1** was also operated in 98% H₂¹⁸O. Isotopically labeled ¹⁸O₂ was detected by gas chromatography-mass spectrometry (GC-MS) after the PEC reaction, and this result confirms that water was the oxygen source (Fig. S4†).

During the PEC reaction for 1 h (Fig. 6c), 4 $\mu\text{mol O}_2$ were produced by TiO₂|TPA|PNO|**1**, corresponding to a TON of approximately 80, while 1.2 $\mu\text{mol O}_2$ was produced by TiO₂|TPA|**1** with a TON of 13 (see the ESI† for experimental details). The higher TON value obtained by TiO₂|TPA|PNO|**1** is in line with the superior photocurrent response and stability of TiO₂|TPA|PNO|**1** over TiO₂|TPA|**1**. The differences exhibited by these two photoanodes demonstrate the value of the redox mediator in a multi-step electron transfer reaction. As shown in Table 1, the remarkable PEC performance achieved by TiO₂|TPA|PNO|**1** is superior to those of other reported photoanodes based on organic dyes in terms of photocurrent and stability. The unique role of PNO was further highlighted by replacing the PNO interlayer with PMMA or Nafion, and the resulting samples (TiO₂|TPA|PMMA|**1** and TiO₂|TPA|Nafion|**1**) afforded even lower photocurrent densities than that of an interlayer-free electrode, TiO₂|TPA|**1** (Fig. S5 and S6†).

The wavelength dependence of the incident photon-to-current conversion efficiency (IPCE) for TiO₂|TPA|PNO|**1** was evaluated at 0.4 V vs. NHE (Fig. 6d). At the visible absorption maximum of TPA (Fig. S7†), an IPCE of 30% was achieved, and the value is comparable to the highest value documented for DSPEC photoanodes.¹⁸ Based on the above measurements, a 1 sun photocurrent density of 500 $\mu\text{A cm}^{-2}$ was estimated by integrating the IPCE curves over the AM 1.5G solar spectrum (Fig. S8†). This result agrees well with the experimental data in Fig. 6a, indicating the reliability of our PEC tests.

The CV of TPA dye in Fig. 7a shows a ground state potential for TPA^{+•/0} at 1.2 V vs. NHE. Based on eqn (2), the excited-state potential for TPA^{+•/0*} on TiO₂ was estimated to be -1.9 V vs. NHE in neutral phosphate buffer,^{39–41} where k_B is the Boltzmann constant, F is the Faraday constant and T is the background temperature. The values for E_0 and $\Delta\tilde{\nu}_{0,1/2}$ were obtained from the spectral fitting described in the ESI (Fig. S9†).

$$(TPA^{+•/0*}) = E_{1/2}(TPA^{+•/0}) - \frac{E_0 + \frac{(\Delta\tilde{\nu}_{0,1/2})^2}{16k_B T \ln(2)}}{F} \quad (2)$$

The finding of a more negative potential for the excited-state of TPA dye than that of the TiO₂ conduction band (-0.55 V vs. NHE at pH 7) is a direct indication of the ability of excited TPA

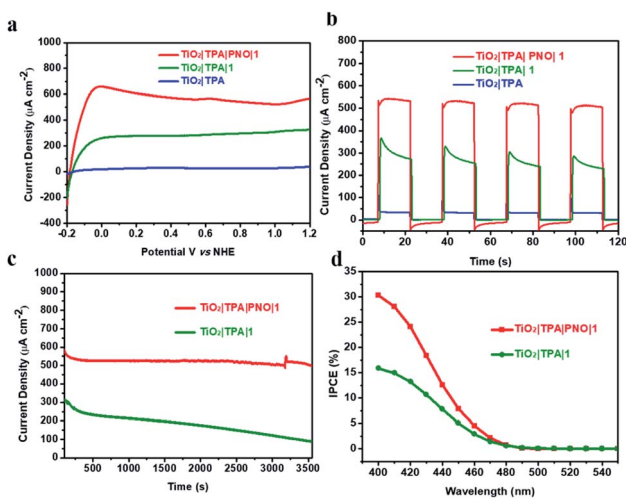


Fig. 6 (a) Linear sweep voltammograms for TiO₂|TPA|PNO|**1** (red), TiO₂|TPA|**1** (green) and TiO₂|TPA (blue) under 1 sun illumination in pH = 7, 0.1 M phosphate buffer containing 0.4 M NaClO₄. (b) Photocurrent density–time traces over 15 s dark–light cycles for the three photoanodes at a constant bias of 0.4 V vs. NHE. (c) Photocurrents for TiO₂|TPA|PNO|**1** (red) and TiO₂|TPA|**1** (green) at a constant bias of 0.4 V vs. NHE under continuous 1 sun illumination. (d) Incident photon-to-current efficiencies (IPCEs) for TiO₂|TPA|PNO|**1** (red) and TiO₂|TPA|**1** (green) at 0.4 V vs. NHE.



Table 1 Performance for photoelectrochemical water oxidation with organic dyes

Electrode	Photocurrent density ^a ($\mu\text{A cm}^{-2}$)	Duration	Conditions	Ref.
$\text{SnO}_2 \text{PMPDI} \text{CoO}_x$	From ~ 40 to ~ 20	300 s	0.4 V vs. NHE, AM 1.5, pH 7	9
$\text{TiO}_2 \text{Al}_2\text{O}_3 \text{PMI} \text{Ir-Sil}$	From ~ 75 to ~ 20	600 s	0.51 V vs. NHE, 200 mW cm^{-2} , $\lambda > 410 \text{ nm}$, pH 5.8	10
$\text{TiO}_2 \text{porphyrin} \text{Cp}^*-\text{Ir}$	~ 60	60 s	0.5 V vs. NHE, 200 mW cm^{-2} , $\lambda > 400 \text{ nm}$, pH 7	11
$\text{TiO}_2 \text{porphyrins} \text{IrO}_2$	From ~ 120 to ~ 30	100 s	0.30 V vs. NHE, AM 1.5, $\lambda > 410 \text{ nm}$, pH 6.8	13
$\text{TiO}_2 \text{subporphyrin} \text{Ru}(\text{bda})$	From ~ 150 to ~ 70	30 s	0.2 V vs. NHE, 100 mW cm^{-2} , $\lambda > 420 \text{ nm}$, pH 7.3	15
$\text{TiO}_2 \text{porphyrin} \text{Ru}(\text{bda})$	From ~ 130 to ~ 80	60 s	-0.2 V vs. NHE , 35 mW cm^{-2} , $\lambda > 380 \text{ nm}$, pH 7.3	16
$\text{TiO}_2 \text{JG}^b \text{Prussian blue}$	From ~ 60 to ~ 45	1 h	0.8 V vs. NHE, 100 mW cm^{-2} , $\lambda > 420 \text{ nm}$, pH 7	17
$\text{SnO}_2/\text{TiO}_2 \text{TPA} \text{Ru}(\text{bda})$	From ~ 200 to ~ 50	1 h	0.4 V vs. NHE, 100 mW cm^{-2} , $\lambda > 400 \text{ nm}$, pH 4.8	18
$\text{SnO}_2/\text{TiO}_2 \text{TPA} \text{Ru}(\text{bda})$	From ~ 200 to ~ 100	60 s	0.45 V vs. NHE, 100 mW cm^{-2} , $\lambda > 400 \text{ nm}$, pH 7	21
$\text{TiO}_2 \text{TPA} \text{PNO} \text{Ru}(\text{bda})$	~ 520	1 h	0.4 V vs. NHE, 100 mW cm^{-2} , $\lambda > 400 \text{ nm}$, pH 7	This work

^a Data were extracted from current density–time traces in the literature. ^b JG = Janus green B.



Fig. 7 (a) Cyclic voltammograms of nanoITO|TPA in pH 7, 0.1 M phosphate buffer (top) and catalyst **1** (1 mM) in mixed $\text{CH}_3\text{CN}/\text{H}_2\text{O}$ ($v/v = 1 : 2$) solution (bottom) with a scan rate of 20 mV s^{-1} . (b) Schematic energy diagram for $\text{TiO}_2|\text{TPA}|\text{PNO}|\mathbf{1}$.

to inject electrons into the TiO_2 underlayer. Since electrochemical water oxidation by **1** on TiO_2 or nanoITO is kinetically prohibited by the presence of long alkyl chains (Fig. S10†), CV of Ru-bda was carried out in homogeneous solution and an onset potential of 0.9 V vs. NHE was determined (Fig. 7a). Bringing these data together, the desired electron flow through spatially arranged redox active components is illustrated in Fig. 7b. The reaction scheme for the activation of the surface assembly and charge separation is shown in eqn (3)–(5). In the assembly, PNO provides a suitable redox potential to establish a free energy gradient allowing electron transfer towards the conducting oxide and hole transfer away from the surface.



In order to fully optimise the performance of hybrid photoanodes, a macrocyclic Ru-bda derivative (**2**)^{42,43} and a Ru-tda derivative (**3**) ($\text{H}_2\text{tda} = [2,2':6',2''\text{-terpyridine}]-6,6''\text{-dicarboxylic acid}$)^{44,45} were selected for catalyst screening (Fig. 2). Both types have shown excellent activity in either electrochemical or

photochemical water oxidation. Among them, hydrophobic catalyst **2** was able to be stably immobilized on $\text{TiO}_2|\text{TPA}$ by drop-casting. Akin to catalyst **1**, C11 alkyl chains were functionalized on Ru-tda (**3**) to facilitate its loading on an electrode surface. The performance of $\text{TiO}_2|\text{TPA}|\text{PNO}|\mathbf{2}$ and $\text{TiO}_2|\text{TPA}|\text{PNO}|\mathbf{3}$ at a catalyst loading of $\Gamma \approx 5 \times 10^{-8} \text{ mol cm}^{-2}$ was examined under the same experimental conditions used for **1**. However, both catalysts showed unsatisfactory catalytic performance (Fig. 8). Particularly, $\text{TiO}_2|\text{TPA}|\text{PNO}|\mathbf{3}$ produced a photocurrent lower than its analog $\text{TiO}_2|\text{TPA}|\mathbf{3}$, probably due to the unfavorable thermodynamics (the catalytic onset potential of **3** is more positive than the redox potential of the NPO couple, Fig. S11†).

The photophysical events that occur upon light absorption are a multi-step electron transfer process, wherein the desired reactions are in kinetic competition with undesired recombination reactions (eqn (6)). To unravel the kinetics of interfacial charge separation and recombination at the photoanode, open-circuit photovoltage decay (OCVAD) was employed as a powerful tool.⁴⁶ This technique has been used for dye-sensitized solar cells (DSSCs) to monitor the transient of open-circuit photovoltage (V_{oc}) during relaxation from the illuminated quasi-equilibrium state to the dark equilibrium.⁴⁷ In a DSPEC, excited-state electron injection occurs at the picosecond timescale, this process is followed by cross-surface electron transfer from a WOC to the oxidized dye, and the latter results in the recovery of most of the oxidized dyes.^{48,49} Under open circuit conditions,



Fig. 8 Linear sweep voltammograms for (a) $\text{TiO}_2|\text{TPA}|\text{PNO}|\mathbf{2}$ (blue) and $\text{TiO}_2|\text{TPA}|\mathbf{2}$, and (b) $\text{TiO}_2|\text{TPA}|\text{PNO}|\mathbf{3}$ (red) and $\text{TiO}_2|\text{TPA}|\mathbf{3}$ (green) under $>400 \text{ nm}$ illumination at 100 mW cm^{-2} in pH 7, 0.1 M phosphate buffer containing 0.4 M NaClO_4 .



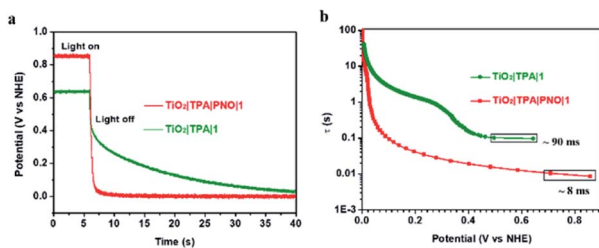


Fig. 9 (a) Open-circuit voltage decays for the electrodes TiO₂-[TPA|PNO]1 (red) and TiO₂|TPA|1 electrodes (green). (b) Electron lifetime derived from eqn (7) plotted as a function of V_{oc} at the transient when illumination (>400 nm, 100 mW cm⁻²) was removed. The results were obtained in 0.1 M phosphate buffer at pH 7.

the free electron density in TiO₂ is roughly determined by electron injection from the photoexcited dye and the back electron transfer of the photogenerated electrons to the oxidized catalysts. Assuming that a majority of the oxidative equivalents are located on the catalyst, the latter step closely resembles the recombination of photogenerated electrons with the oxidized form of the redox couple in DSSCs. Under illumination, more efficient charge separation enables higher free electron density in TiO₂ and a larger V_{oc} . When light is stopped, a faster V_{oc} decay is expected due to enhanced charge recombination driven by the high density of free electrons. Based on the research of Zaban and Bisquert *et al.*,^{50,51} the lifetime of photogenerated electrons (free electron lifetime τ) in the semiconductor nanostructure and V_{oc} comply with eqn (7), where k_B is the Boltzmann constant and T is the temperature.

$$\tau = -\frac{k_B T}{e} \left(\frac{dV_{oc}}{dt} \right)^{-1} \quad (7)$$

To this end, a free electron lifetime of 8 ms for TiO₂-[TPA|PNO]1 was quantified at the transient when illumination is stopped. This value was increased by an order of magnitude in the absence of a PNO layer, as a τ value of 90 ms was obtained by TiO₂|TPA|1 (Fig. 9). Accordingly, more efficient charge separation under illumination was expected for TiO₂-[TPA|PNO]1 than TiO₂|TPA|1. Longer electron lifetimes were also given by Nafion (150 ms) and PMMA (100 ms) coated photoanodes (Fig. S12[†]). The variations in the lifetimes of free electrons are in line with the trend of photocurrent densities achieved by these photoanodes, revealing the crucial role of PNO in facilitating light-induced charge separation at the interface of the photoanode, analogous to tyrosine in PSII.^{52,53}

Conclusions

In summary, we have demonstrated that cheap and easily available pyridine *N*-oxide can serve as an efficient electron transfer relay in a TiO₂ photoanode co-loaded with organic dye and a molecular WOC. As a close mimic of the tyrosine/tyrosyl radical pair in PSII, the PNO interlayer cycles between its reduced form and a radical cation in photolysis and improves light-induced charge separation by forming a suitable electron

transfer gradient between the catalyst and semiconductor oxide. The integrated photoanode exhibited a highly stable photocurrent over 500 $\mu\text{A cm}^{-2}$ that continued for at least 1 hour without obvious decay, which opens a new avenue for rational design of low-cost organic photoanodes. In addition, the hybrid architecture described here was benefited by a facile drop-casting process. The surface assembly organized by hydrophobic interactions overcomes the limitation of a covalent bond between the linker and surface in aqueous solutions and adds an additional advantage to the stability of the photoanode. The observation of an impressive 1 sun photocurrent up to 3 mA cm⁻² with a single organic dye under sacrificial conditions implies that further improvement in the efficiency and stability of organic dye-based PEC cells may be accessible with the assistance of faster water oxidation catalysts.

Data availability

All relevant experimental procedures and characterization data have been reported above and in the ESI.[†]

Author contributions

F. L. and L. S. supervised this project; Y. Z., X. L. and F. L. designed the experiments; Y. Z., G. L., R. Z. and H. G. prepared the electrodes, and performed characterization and photoelectrochemical measurements; Y. Z. and F. L. wrote the manuscript.

Conflicts of interest

The authors declare no competing financial interest.

Acknowledgements

This work was supported by the National Natural Science Foundation of China (22088102, 21872016, and 22172018), the Fundamental Research Funds for the Central Universities of China (DUT20LAB307) and the K&A Wallenberg Foundation.

References

- 1 F. Li, H. Yang, W. Li and L. Sun, *Joule*, 2018, **2**, 36–60.
- 2 B. Zhang and L. Sun, *Chem. Soc. Rev.*, 2019, **48**, 2216–2264.
- 3 B. Zhang and L. Sun, *J. Am. Chem. Soc.*, 2019, **141**, 5565–5580.
- 4 M. K. Brennaman, R. J. Dillon, L. Alibabaei, M. K. Gish, C. J. Dares, D. L. Ashford, R. L. House, G. J. Meyer, J. M. Papanikolas and T. J. Meyer, *J. Am. Chem. Soc.*, 2016, **138**, 13085–13102.
- 5 J. Li, Y. Zhu, F. Li, G. Liu, S. Xu and L. Sun, *Chin. J. Catal.*, 2021, **42**, 1352–1359.
- 6 J. R. Swierk, N. S. McCool and T. E. Mallouk, *J. Phys. Chem. C*, 2015, **119**, 13858–13867.
- 7 T. J. Meyer, M. V. Sheridan and B. D. Sherman, *Chem. Soc. Rev.*, 2017, **46**, 6148–6169.
- 8 J. T. Kirner and R. G. Finke, *J. Mater. Chem. A*, 2017, **5**, 19560–19592.



- 9 J. T. Kirner and R. G. Finke, *ACS Appl. Mater. Interfaces*, 2017, **9**, 27625–27637.
- 10 R. J. Kamire, K. L. Materna, W. L. Hoffeditz, B. T. Phelan, J. M. Thomsen, O. K. Farha, J. T. Hupp, G. W. Brudvig and M. R. Wasielewski, *J. Phys. Chem. C*, 2017, **121**, 3752–3764.
- 11 G. F. Moore, J. D. Blakemore, R. L. Milot, J. F. Hull, H. E. Song, L. Cai, C. A. Schmuttenmaer, R. H. Crabtree and G. W. Brudvig, *Energy Environ. Sci.*, 2011, **4**, 2389–2392.
- 12 A. Nayak, R. R. Knauf, K. Hanson, L. Alibabaei, J. J. Concepcion, D. L. Ashford, J. L. Dempsey and T. J. Meyer, *Chem. Sci.*, 2014, **5**, 3115–3119.
- 13 J. R. Swierk, D. D. Méndez-hernández, N. S. McCool, P. Liddell, Y. Terazono, I. Pahk, J. J. Tomlin, N. V. Oster, T. A. Moore, A. L. Moore, D. Gust and T. E. Mallouk, *Proc. Natl. Acad. Sci. U. S. A.*, 2015, **112**, 1681–1686.
- 14 P. K. Poddutoori, J. M. Thomsen, R. L. Milot, S. W. Sheehan, C. F. A. Negre, V. K. R. Garapati, C. A. Schmuttenmaer, V. S. Batista, G. W. Brudvig and A. Van Der Est, *J. Mater. Chem. A*, 2015, **3**, 3868–3879.
- 15 M. Yamamoto, Y. Nishizawa, P. Chábera, F. Li, T. Pascher, V. Sundström, L. Sun and H. Imahori, *Chem. Commun.*, 2016, **52**, 13702–13705.
- 16 M. Yamamoto, L. Wang, F. Li, T. Fukushima, K. Tanaka, L. Sun and H. Imahori, *Chem. Sci.*, 2016, **7**, 1430–1439.
- 17 T. G. Ulusoy Ghobadi, A. Ghobadi, M. Buyuktemiz, E. A. Yildiz, D. Berna Yildiz, H. G. Yaglioglu, Y. Dede, E. Ozbay and F. Karadas, *Angew. Chem., Int. Ed.*, 2020, **59**, 4082–4090.
- 18 M. S. Eberhart, D. Wang, R. N. Sampaio, S. L. Marquard, B. Shan, M. K. Brennaman, G. J. Meyer, C. Dares and T. J. Meyer, *J. Am. Chem. Soc.*, 2017, **139**, 16248–16255.
- 19 F. Li, K. Fan, B. Xu, E. Gabrielsson, Q. Daniel, L. Li and L. Sun, *J. Am. Chem. Soc.*, 2015, **137**, 9153–9159.
- 20 Y. K. Eom, L. Nhon, G. Leem, B. D. Sherman, D. Wang, L. Troian-Gautier, S. Kim, J. Kim, T. J. Meyer, J. R. Reynolds and K. S. Schanze, *ACS Energy Lett.*, 2018, **3**, 2114–2119.
- 21 K. R. Wee, B. D. Sherman, M. K. Brennaman, M. V. Sheridan, A. Nayak, L. Alibabaei and T. J. Meyer, *J. Mater. Chem. A*, 2016, **4**, 2969–2975.
- 22 Y. Zhu, D. Wang, Q. Huang, J. Du, L. Sun, F. Li and T. J. Meyer, *Nat. Commun.*, 2020, **11**, 1–8.
- 23 K. Hanson, M. D. Losego, B. Kalanyan, D. L. Ashford, G. N. Parsons and T. J. Meyer, *Chem. Mater.*, 2013, **25**, 3–5.
- 24 L. Li, L. Duan, Y. Xu, M. Gorlov, A. Hagfeldt and L. Sun, *Chem. Commun.*, 2010, **46**, 7307–7309.
- 25 L. Zhang, Y. Gao and X. Ding, *Electrochim. Acta*, 2016, **207**, 130–134.
- 26 K. R. Wee, M. K. Brennaman, L. Alibabaei, B. H. Farnum, B. Sherman, A. M. Lapidés and T. J. Meyer, *J. Am. Chem. Soc.*, 2014, **136**, 13514–13517.
- 27 Y. Umena, K. Kawakami, J. R. Shen and N. Kamiya, *Nature*, 2011, **473**, 55–60.
- 28 J. Z. Zhang and E. Reisner, *Nat. Rev. Chem.*, 2020, **4**, 6–21.
- 29 D. Wang, R. N. Sampaio, L. Troian-Gautier, S. L. Marquard, B. H. Farnum, B. D. Sherman, M. V. Sheridan, C. J. Dares, G. J. Meyer and T. J. Meyer, *J. Am. Chem. Soc.*, 2019, **141**, 7926–7933.
- 30 T. Irebo, M. T. Zhang, T. F. Markle, A. M. Scott and L. Hammarström, *J. Am. Chem. Soc.*, 2012, **134**, 16247–16254.
- 31 J. M. Keough, A. N. Zuniga, D. L. Jenson and B. A. Barry, *J. Phys. Chem. B*, 2013, **117**, 1296–1307.
- 32 Y. Zhao, J. R. Swierk, J. D. Megiatto, B. Sherman, W. J. Youngblood, D. Qin, D. M. Lentz, A. L. Moore, T. A. Moore, D. Gust and T. E. Mallouk, *Proc. Natl. Acad. Sci. U. S. A.*, 2012, **109**, 15612–15616.
- 33 D. Wang, F. Niu, M. J. Mortelliti, M. V. Sheridan, B. D. Sherman, Y. Zhu, J. R. McBride, J. L. Dempsey, S. Shen, C. J. Dares, F. Li and T. J. Meyer, *Proc. Natl. Acad. Sci. U. S. A.*, 2020, **117**, 12564–12571.
- 34 L. Zhang, X. Yang, W. Wang, G. G. Gurzadyan, J. Li, X. Li, J. An, Z. Yu, H. Wang, B. Cai, A. Hagfeldt and L. Sun, *ACS Energy Lett.*, 2019, **4**, 943–951.
- 35 Y. Wang, F. Li, X. Zhou, F. Yu, J. Du, L. Bai and L. Sun, *Angew. Chem., Int. Ed.*, 2017, **56**, 6911–6915.
- 36 S. I. Kulakovskaya, S. N. Shamaev and V. M. Berdnikov, *Bull. Acad. Sci. USSR, Div. Chem. Sci.*, 1990, **39**, 1345–1348.
- 37 H. Li, F. Xie and M. T. Zhang, *ACS Catal.*, 2021, **11**, 68–73.
- 38 D. Wang, Z. Xu, M. Sheridan, J. Concepcion, F. Li, T. Lian and T. J. Meyer, *Chem. Sci.*, 2021, **12**, 14441–14450.
- 39 J. P. Claude and T. J. Meyer, *J. Phys. Chem.*, 1995, **99**, 51–54.
- 40 A. Ito and T. J. Meyer, *Phys. Chem. Chem. Phys.*, 2012, **14**, 13731–13745.
- 41 E. M. Kober, J. V. Caspar, R. S. Lumpkin and T. J. Meyer, *J. Phys. Chem.*, 1986, **90**, 3722–3734.
- 42 M. Schulze, V. Kunz, P. D. Frischmann and F. Würthner, *Nat. Chem.*, 2016, **8**, 576–583.
- 43 V. Kunz, J. O. Lindner, M. Schulze, M. I. S. Röhr, D. Schmidt, R. Mitrić and F. Würthner, *Energy Environ. Sci.*, 2017, **10**, 2137–2153.
- 44 R. Matheu, M. Z. Ertem, J. Benet-Buchholz, E. Coronado, V. S. Batista, X. Sala and A. Llobet, *J. Am. Chem. Soc.*, 2015, **137**, 10786–10795.
- 45 R. Matheu, M. Z. Ertem, C. Gimbert-Suriñach, J. Benet-Buchholz, X. Sala and A. Llobet, *ACS Catal.*, 2017, **7**, 6525–6532.
- 46 M. Zhong, T. Hisatomi, Y. Kuang, J. Zhao, M. Liu, A. Iwase, Q. Jia, H. Nishiyama, T. Minegishi, M. Nakabayashi, N. Shibata, R. Niishiro, C. Katayama, H. Shibano, M. Katayama, A. Kudo, T. Yamada and K. Domen, *J. Am. Chem. Soc.*, 2015, **137**, 5053–5060.
- 47 Y. K. Eom, I. T. Choi, S. H. Kang, J. Lee, J. Kim, M. J. Ju and H. K. Kim, *Adv. Energy Mater.*, 2015, **5**, 1–9.
- 48 J. R. Swierk, N. S. McCool, T. P. Saunders, G. D. Barber, M. E. Strayer, N. M. Vargas-Barbosa and T. E. Mallouk, *J. Phys. Chem. C*, 2014, **118**, 17046–17053.
- 49 P. Xu, C. L. Gray, L. Xiao and T. E. Mallouk, *J. Am. Chem. Soc.*, 2018, **140**, 11647–11654.
- 50 J. Bisquert, A. Zaban, M. Greenshtein and I. Mora-Seró, *J. Am. Chem. Soc.*, 2004, **126**, 13550–13559.
- 51 A. Zaban, M. Greenshtein and J. Bisquert, *ChemPhysChem*, 2003, **4**, 859–864.
- 52 H. Iwami, M. Okamura, M. Kondo and S. Masaoka, *Angew. Chem.*, 2021, **133**, 6030–6034.
- 53 H. Iwami, M. Kondo and S. Masaoka, *ChemElectroChem*, 2022, **9**, 52–58.

