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Introduction

Organic photovoltaic cells (OPVs) have attracted enormous attention as a promising alternative to conventional fossil resources for its unique advantages, including low cost, light weight, mechanical flexibility and solution processability.¹ Moreover, solution processability allows the low-cost fabrication of large-area by rollto-roll (R2R) printing, in contrast to traditional Si-based solar cells. Currently, power conversion efficiencies (PCEs) over 10% have been achieved for both polymer and small-molecule based solar cells with bulk-heterojunction (BHJ) architecture.²⁻⁶ Organic solar cells are sandwich structures including a photoactive layer, interfacial buffer layers, and electrodes. Up till now, the advancement of record-high PCEs are mainly driven by the synthesis of novel electron-donor polymers⁷⁻¹⁰ and small molecules,¹¹⁻¹⁴ and electronacceptor materials,¹⁵⁻¹⁷ along with devices optimization processing.¹¹,

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Investigation for the enhanced performance and lifetime of organic solar cells using solution-processed carbon dots as the electron transport layer

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Easily prepared and stable solution-processed carbon dots (CDs) have been used and systematically investigated as the electron transport layers (ETLs) for both small-molecule and polymer based solar cells. Significantly enhanced device performance and lifetime are observed. The enhanced performance is mainly driven by the improvements of the short circuit current (J_{sc}) and the fill factor (FF), caused by decreasing the work function of Al electrode and series resistance, increasing shunt resistances, and balancing electron and hole mobility. Therefore, the devices with CDs as the ETLs have higher charge transport and collection efficiency. In addition, lifetimes of the devices with CDs as the ETLs are also significantly improved, due to the much better air-stability of CD material compared to LiF as the ETLs. And another reason is that it can efficiently prevent the forming of unstable cathode contact for the diffusion of Al ions at the interface. These results indicate that CD, a relative cheap and stable material, has great potential to be a promising ETL material for industrial-scale manufacture of organic solar cells.

> 12, 18, 19, 20 Therefore, interfacial engineering also plays a critical role in determining the performance of OPVs.^{21, 22} Better interfacial layers contribute to providing better energy alignment, minimizing contact resistance and realizing ohmic contacts between the active layer and the electrodes for better charge extraction.²³ Furthermore, the optimized interfacial layers could also prevent the recombination of electron and hole by improving of the charge transport and extraction pathways.²⁴ So far, LiF²⁵ and Ca²⁶ as the ETLs have been widely used in OPVs. However, the thickness and evaporation rate of those ETLs are difficult to control, which can also cause significant cost to the fabrication of large-area. Besides, these ETL materials are very sensitive to moisture and oxygen, causing the inferior device stability and operating lifetime. Thus, for future possible industry applications, solution-processed ETLs with low synthesis cost, and high devices performances repeatability and stability are highly needed. Recently, various solution-processed ETLs, such as nanoscale transition metal oxides (e.g. ZnO),²⁷ fullerene derivatives^{28, 29} and polyelectrolytes(e.g. PEI, PFN)^{22, 30} have been studied and in some cases give exciting performances. However, these materials in most cases, including the nano ZnO, are difficult to prepare and/or very expensive. And some previous studies based on graphene quantum dots (GQDs) as hole extraction layers have been reproted in polymer solar cells, and GQDs could improve the lifetime and improving the contact between the blended polymer layer and anode.31, 32, 33

> Herein, carbon dots (CDs), easily synthesized from low-cost commercial off-the-shelf materials by a facile and efficient one-step strategy, were studied as the ETL for both small-molecule solar cells

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[†] Electronic supplementary information (ESI) available: General characterization, UV-vis, FT-IR,PL, XPS, TEM and XRD spectra of the CDs, AFM, SCLC, UPS figures, etc. See DOI: 10.1039/b000000x/

ETL	V _{oc} (V)	$J_{\rm sc}$ (mA cm ⁻²)	FF (%)	PCE ^a (%)	$\frac{R_{\rm s}}{(\Omega \ {\rm cm}^2)}$	$\frac{R_{\rm sh}}{(\Omega \ {\rm cm}^2)}$	$(\mathrm{cm}^2 \mathrm{V}^{-1} \mathrm{s}^{-1})$
None ^b	0.895	12.88	60.8	7.00(7.19)	13.1	716.9	4.39×10 ⁻⁴
LiF ^c	0.910	12.92	61.8	7.22(7.51)	11.7	1134.7	3.44×10 ⁻⁴
1.0mg/ml ^d	0.903	13.02	61.3	7.20(7.33)	11.2	428.5	2.32×10 ⁻⁴
$0.5 mg/ml^d$	0.903	13.01	61.9	7.27(7.39)	9.4	1454.1	2.62×10 ⁻⁴
0.1mg/ml ^d	0.904	13.32	63.7	7.67(7.78)	9.3	1577.1	3.06×10 ⁻⁴
0.05mg/ml^{d}	0.905	13.12	62.8	7.45(7.52)	11.5	1123.0	3.41×10 ⁻⁴
0.01mg/ml^{d}	0.904	13.03	61.4	7.23(7.48)	12.1	1104.2	4.30×10 ⁻⁴

Table 1 Average OPV Performance Parameters for DR3TBDTT:PC₇₁BM (1:0.8w/w) based solar cells with the various ETLs (device structure, ITO/PEDOT:PSS/DR3TBDTT:PC₇₁BM /ETL/Al), along with the shunt resistance (R_{sh}) and series resistance (R_{s}) and electron mobility.

^aThe best efficiency is shown in parentheses. ^bThe devices without ETL. ^cThe devices with LiF as the ETL, the data is from Ref 11. ^dThe devices with CDs of various concentrations as the ETL. All the data are based on the results of 15 devices.



Scheme 1 Molecular structures of a) DR3TBDTT, b) $PC_{71}BM$, c) P3HT, and d) $PC_{61}BM$.

(SMSCs) and polymer solar cells (PSCs) with the blend of small molecule DR3TBDTT:[6,6]-phenyl- C_{71} -butyric acid methyl ester(DR3TBDTT:PC₇₁M) and poly(3-hexylthiophene): [6,6]-phenyl- C_{61} -butyric acid methyl ester (P3HT:PC₆₁BM) as the active layer, respectively. As a result, significantly improved both performance and lifetimes were observed. Furthermore, the cells with CDs as the ETLs exhibited excellent reproducibility using an environmentally friendly solvent (methanol).

Results and Discussion

CDs synthesis and characterization

CDs were prepared following the literature³⁴ (see Experimental Section for the details) with slight modification, and characterized using various spectroscopic methods (see Fig. S1-S6). Fig. S3 shows the fluorescence spectra of the CDs, which exhibits an excitation-dependent emission phenomenon. The TEM image shows that our CDs are of uniform sizes and have a narrow distribution in the range of 1-2 nm (Fig. S5).

CDs as the ETLs for small molecule based device

The photovoltaic performances of small-molecule solar cells with CDs as the ETL prepared from CD methanol solutions with different concentrations, together with the control devices, are systematically investigated. The thickness of CD based ETL can be easily controlled by using different concentrations of CDs solution and partially cover the active layer (see Fig. S7 and Table S1 for the details). The cell structure is ITO/PEDOT:PSS/DR3TBDTT:PC71BM/ETLs/Al. The molecular structures of DR3TBDTT and PC71BM are displayed in Scheme 1a and 1b, respectively. The average device performance parameters (based on min of 30 devices) with CDs of various concentrations and the control/compared devices without ETL/with LiF as the ETL under the illumination of AM 1.5G, 100 mW cm⁻² are summarized in Table 1. Typical current density versus voltage (J - V) curves of the devices are shown in Fig. 1a. The device without an ETL exhibits an average PCE of 7.00% with V_{oc} of 0.895 V, J_{sc} of 12.88 mA cm⁻², and FF of 60.8%. The device with LiF as the ETL exhibits an average PCE of 7.22% with V_{oc} of 0.91 V, J_{sc} of 12.92 mA cm⁻², and FF of 61.8%.¹¹ Notably, using solution-processed CDs with various concentrations as the ETLs, the devices clearly exhibit enhanced performance compared to that of the device without ETL layer and more importantly that of the device with LiF as the ETL. The device with CDs of the optimal concentration (0.1 mg/ml) as the ETL exhibits an average PCE of 7.67% with V_{oc} of 0.904 V, J_{sc} of 13.32 mA cm⁻², and FF of 63.7%. From Table 1, the series resistance (R_s) decreases from 13.1 to 11.7 and 9.3 Ω cm⁻², and shunt resistances $(R_{\rm sh})$ increases significantly from 716.9 to 1134.7 and 1577.1 Ω cm⁻² for the device without ETL, the device with LiF as the ETL, and the device with CDs (0.1 mg/ml) as the ETL, respectively. Thus, the reduced $R_{\rm s}$ and increased $R_{\rm sh}$ are expected to improve the ohmic contact and reduce the leakage current (Fig. S8), and therefore, higher FFs are observed with CD as the ETL. In addition, the electron mobility measurements based on the space-charge limited current (SCLC) model with the device structure Al/DR3TBDTT:PC71BM/ETLs/Al are shown in Fig. S9 and the



Fig. 1 (a) Current density versus voltage (J - V) curves and (b) External quantum efficiency (EQE) curves characteristics for the devices without ETL (squares), with LiF as the ETL (circles), and with CDs of the optimal concentration as the ETL (up-triangles, 0.1 mg/ml). All the devices under simulated 100 mW cm⁻² AM 1.5G illumination.

results are also summarized in Table 1.The hole mobility¹¹ of 2.88×10^{-4} cm² V⁻¹ s⁻¹ was measured with the device structure ITO/PEDOT:PSS/DR3TBDTT:PC₇₁BM/MoO₃/Au. The electron mobility for the device with CDs (0.1 mg/ml) was measured with a value of 3.06×10^{-4} cm² V⁻¹ s⁻¹. Thus, more balanced charge transport in the device is achieved. This may increase the FF by restricting the build-up of space charges, and hence, reducing charge recombination.

The external quantum efficiency (EQE) spectra of the DR3TBDTT based devices with CD as the ETL and the compared devices are investigated as shown in Fig. 1b. The EQE curve for the device with CDs (0.1 mg/ml) as the ETL exhibits better response than those of devices without ETL, and with LiF as the ETL. Especially, the EQE curve for the device with CDs (0.1 mg/ml) exceeds 70% in a wide range of 450 – 600 nm with the maximum value of over 75% at 490 nm. In comparison, the device without ETL, and the device with LiF as the ETL show peaks of 71% at 470 nm, and 74% at 510 nm, respectively. All the calculated J_{sc} values obtained by integration of the EQE data agree well with the J_{sc} values from the J - V measurements. Those results prove that CDs as the ETL could indeed improve the J_{sc} .

The increased J_{sc} in the OPV devices may originate from reduced bimolecular recombination, the increased absorption of photons, or a combination of both.³⁵ To gain deeper insight into the influence of



Fig. 2 (a) Photocurrent density versus effective voltage $(J_{\rm ph} - V_{\rm eff})$ and (b)Correlation $J_{\rm ph,se}$ / $J_{\rm ph,sat}$ and $J_{\rm ph,m}$ / $J_{\rm ph,sat}$ characteristics for the devices without ETL (squares), with LiF as the ETL (circles), and with CDs of the optimal concentration as the ETL (up-triangles, 0.1 mg/ml). All the devices under constant incident light intensity simulated 100 mW cm⁻² AM 1.5G illumination.

the CDs buffer layers on the device performances, photocurrent density versus effective voltage $(J_{ph} - V_{eff})$ characteristics are investigated for the devices with or without ETLs under constant incident light intensity (AM 1.5G, 100 mW cm⁻²). $J_{\rm ph} = J_{\rm L} - J_{\rm D}$, where $J_{\rm L}$ and $J_{\rm D}$ are the current densities under illumination and in the dark, respectively. For $V_{\text{eff}} = V_0 - V_{\text{appl}}$, V_0 is the voltage at $J_{\text{ph}} = 0$ and V_{appl} is the applied voltage.^{12, 22}A plot of J_{ph} versus V_{eff} in a wide reverse-bias range is presented in Fig. 2a. Notably, $J_{\rm ph}$ reaches saturation for these devices at a large reverse voltage (e.g. V_{eff} = 2.5 V, see Fig. S10 for the detailed J_{ph} - V_{eff} without the log graph), which suggests that at this voltage the photogenerated excitons are dissociated into free carriers and carriers are collected by the electrodes with little geminate or bimolecular recombination. The ratio $J_{\rm ph}/J_{\rm sat}$ can be used to judge the overall exciton dissociation efficiency and charge collection efficiency.³⁶ Under short-circuit conditions, the ratios J_{ph.sc}/J_{ph.sat} are 92.4%, 93.4%, and 96.1% for the device without ETL, the device with LiF as the ETL, and the device with CDs (0.1 mg/ml) as the ETL, respectively. At the maximal power output conditions, for the device without ETL, the device with LiF as the ETL, and the device with CDs (0.1 mg/ml) as the ETL, J_{ph,m}/J_{ph,sat} are 78.3%, 77.5%, and 81.3%, respectively. Those results indicate that the device with CDs (0.1 mg/ml) as the ETL has higher exciton dissociation and charge collection efficiency

ETL	V _{oc} (V)	$J_{\rm sc}$ (mA cm ⁻²)	FF (%)	PCE ^a (%)	$R_{\rm s}$ $\Omega {\rm cm}^{-2}$	$R_{\rm sh}$ $\Omega {\rm cm}^{-2}$
None ^b	0.610	9.98	50.4	3.06(3.08)	14.6	373.0
LiF ^c	0.604	10.03	55.8	3.38(3.46)	13.2	643.2
1.0 mg/ml ^d	0.609	10.11	51.7	3.18(3.29)	13.5	623.8
0.5 mg/ml ^d	0.609	10.25	54.8	3.42(3.52)	12.3	691.6
0.1 mg/ml ^d	0.608	10.13	52.8	3.25(3.34)	11.7	694.1
0.05 mg/ml ^d	0.613	9.86	0.521	3.15(3.21)	13.4	631.2

Table 2 Average OPV Performance Parameters for bulk heterojunction (BHJ) solar cells based on P3HT:PC₆₁BM (1:1w/w) with or without ETLs, along with the shunt resistance (R_{sh}) and series resistance (R_{sh}).

^aThe best efficiency is shown in parentheses. ^bThe devices without ETL. ^cThe reference devices with LiF as the ETL. ^dThe devices with CDs of various concentrations as the ETL. All the data are based on the results from 15 devices.

(Fig. 2b). Though the overall performance is limited by using CD as the ETL compared with that when using LiF as the ETL, the superior $J_{\rm ph} - V_{\rm eff}$ characteristics of devices clearly demonstrate that CDs as the ETLs can reduce bimolecular recombination and increase charge collection efficiency. In addition, a lower work function (-3.56 vs -4.3 eV, Fig. S11 and Table S2)³⁷ is also observed for the Al electrode when CD is used as the ETL. This should generate a higher built-in voltage ($V_{\rm bi}$), which is believed to facilitate the change extraction/separation. Thus, higher $J_{\rm sc}$ of device with CDs as the ETL is observed. And absorption spectra of the blends and CDs (0.1 mg/mL) spin-coated onto the active layer were showed in Fig S12. There was nearly no difference in the absorption. So reduced bimolecular recombination might be the dominant factor.

Enhanced lifetime using CDs as the ETLs

One of the most challenging issues for OPV is their limited lifetime. To investigate the stability and feasibility for large-scale organic solar cells manufacture, lifetime testing experiments of SMSCs with different ETLs are conducted. The stability testing used in this analysis was in accordance with ISOS-D-1 protocols.³⁸ All the devices are simply sealed by ultraviolet-curable resin and without other complicated encapsulation techniques, and are stored in glove box or in air (humidity, 60%; 25°C). Fig. 3 shows PCEs degradation over time for the device without ETL, the device with LiF as the ETL, and the device with CDs (0.1 mg/ml) as the ETL. As shown in Fig. 3, the degradation of photovoltaic performance for the device with CDs (0.1 mg/ml) is much slower than that of the devices without ETL, and with LiF as the ETL in both glove box and air. The PCE of device with LiF as the ETL drops rapidly to 54% of an initial PCE in air after 1970 min (~ 33 h) and 53% of initial PCE in glove box for 10220 min (170 h). In contrast, the device with CDs (0.1 mg/ml) remains PCE near 90% of initial PCE under the same conditions in glove box. And even in air, the device with CDs (0.1 mg/ml) also has PCE over 85% of initial PCE after 11150 min (~ 186 h) (See Fig. S13-S16 for the detailed degradation data in SI). This should be due to the much better air-stability of CD material to



Fig. 3 The normalized efficiency vs storage time for conventional devices without ETL (squares), with LiF as the ETL (circles), and with CDs of the optimal concentration as the ETL (up-triangles, 0.1 mg/ml) stored (a) in glove box and (b) in air (humidity, 60%; 25 ^oC) with simple encapsulation.

both moisture and oxygen, compared to LiF as the ETLs. Another reason may be that CDs as the ETL can efficiently prevent the diffusion of Al ions during the evaporation process. Metal ions formed at the interface tend to migrate into the active layer and form an unstable cathode contact, thus further affect the performance and long-term stability of the devices.^{37, 39} These results indicate that the

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devices with CDs (0.1 mg/ml) as the ETL have a much improved operational stability.

In addition, Table S3 exhibits a statistical analysis of 15 high performance devices for each ETL, showing the reproducible differences for $V_{\rm oc}$, $J_{\rm sc}$, FF and PCE. The photovoltaic parameters of CDs (0.1 mg/ml) as the ETL show overall significantly lower standard deviations (σ), confirming that the cells exhibit better reproducibility for cell-performance.

CDs as the ETLs layer for polymer based devices

To further test the applicability of CDs in OPV cells, we investigated the CDs as the ETLs in polymer solar cells using the landmark material poly(3-hexylthiophene):phenyl-C₆₁-butyric acid methyl ester (P3HT:PC₆₁BM) as the donor. The average device performance parameters with CDs of the various concentrations and the control devices without ETL under the illumination of AM 1.5G, 100 mW cm^{-2} are summarized in Table 2. Typical J - V curves of the optimized devices shown in Fig. S17 indicate the CD as the ETL has similar performance as that for the small molecule based devices studied above. For example, the device without ETL exhibits an average PCE of 3.06% (best PCE 3.08%) with V_{oc} of 0.610 V, FF of 50.4%, and $J_{\rm sc}$ of 9.98 mA cm⁻². The device with LiF as the ETL exhibits an average PCE of 3.38% with V_{oc} of 0.604 V, J_{sc} of 10.03 mA cm⁻², and FF of 55.8%. Notably, The devices with CDs (0.5 mg/ml) gives average PCE of 3.42% (best PCE 3.52%) with V_{oc} of 0.609 V, FF of 54.8%, and $J_{\rm sc}$ of 9.98 mA cm⁻². These results indicate the wide applicability of CDs as the ETLs materials.

Conclusions

In conclusion, an easily made solution-processed CDs has been synthesized and investigated as the ETLs for both small-molecule and polymer based solar cell devices and much improved performances have achieved in terms of both power conversion efficiency and life stability. The enhanced PCEs are due to the improvement of J_{sc} and FF, mainly due to the reduced series resistance, increased shunt resistance, together with the lowed work function of the electrode. In addition, the device lifetime is also improved significantly, caused by the much better air-stability of CD material compared to LiF as the ETLs. Another reason may be that CDs as the ETL can efficiently prevent the forming of unstable cathode contact for the diffusion of Al ions at the interface into the active layer. Compared with the currently widely used ETL materials such as LiF and Ca, these results demonstrate that the easily and cheap and stable carbon dot could be used as an efficient ETL material for both small-molecule and polymer based OPV devices, which might hold some significance for future commercialization of OPV devices.

Experimental Section

Materials Preparation.

DR3TBDTT was synthesized using the method reported previously.¹¹ P3HT was purchased from Rieke Metals, Inc. PC₇₁BM

and $PC_{61}BM$ were purchased from American Dye Source, Inc. All of the materials are used as received unless otherwise specified.

Preparation of carbon dots

Carbon dots were prepared following the literature with slight modification.³⁴ In a typical synthesis, citric acid (2.1014 g) and ethylenediamine (670 μ L) were dissolved in DI-water (30 mL). Then the solution was transferred to a poly (tetrafluoroethylene) (Teflon)-lined autoclave (50 mL) and heated to 250 °C for 10 h. After cooling to room temperature, the reaction solution was filtered through a 0.22 μ m microporous membrane and a brown-black filter solution was obtained, which was further dialyzed in a dialysis bag (retained molecular weight: 1000 Da) for 5 days and the CDs was obtained by vacuum freeze-drying.

Device fabrication

of Devices are fabricated with structure а glass/ITO/PEDOT:PSS/donor:acceptor/ETL/Al. The ITO-coated glass substrates are cleaned by ultrasonic treatment in detergent, deionized water, acetone and, isopropyl alcohol under ultrasonication for 15 min each, and subsequently dry by a nitrogen blow. A thin layer of PEDOT:PSS (Clevios P VP AI 4083, filtered at 0.45 µm) is spin-coated at 4500 r.m.p onto the ITO surface. After being baked at 150 °C for 20 min, the substrates are transferred into an argon-filled glove box. Subsequently, the active layer is spincoated on the HTLs. For the small-molecule solar cells, the chloroform solution containing 10 mg mL⁻¹ of DR3TBDTT and 8.0 mg mL⁻¹ of PC₇₁BM is spin-cast onto the HTLs at 1700 rpm for 20 s. For the polymer solar cells, the o-dichlorobenzene (o-DCB) solution containing 18 mg mL⁻¹ of P3HT and 18 mg mL⁻¹ of PC₆₁BM is spincoated at 800 r.p.m for 18 s. The resulting active-coated substrates are kept in a petri-dish at room temperature for 1 h to allow the o-DCB solvent to evaporate slowly, and then they are annealed inside the glove box at 110^oC for 10 min. Finally, the various concentration of CDs using a methanol solution as ETLs are spin-coated onto the active layer and then 80 nm Al layer is evaporated under high vacuum (< 2×10^{-4} Pa). The effective areas of cells are 4 mm², defined by shadow masks.

Characterization

UV-vis absorption spectra are obtained using a JASCO V-570 spectrometer. Fourier transform infrared (FT-IR) spectra are recorded on a Bruker Tensor 27 spectrometer (Germany). Photoluminescence characterization is done using a Fluoro Max-P luminescence spectrometer using a xenon lamp as the source of excitation. X-ray photoelectron spectroscopy (XPS) analysis is done using an AXIS HIS 165 spectrometer (Kratos Analytical) with a monochromatized Al K α X-ray source (1486.71 eV photons). X-ray diffraction (XRD) experiment is performed on a Bruker D8 FOCUS X-ray diffractometer with Cu K α radiation (k = 1.5406 Å) at a generator voltage of 40 kV and a current of 40 mA. Transmission electron microscopy (TEM) is performed on a Philips Technical G² F20 at 200 kV. Atomic force microscope (AFM) investigation is

performed using a Bruker Multi Mode 8 instrument in the "tapping" mode. The Ultraviolet Photoelectron Spectroscopy (UPS) measurements (Thermo ESCALAB 250) are carried out using the He I (hv = 21.2 eV) source.

The J-V curves for the photovoltaic devices are obtained using a Keithley 2400 source-measure unit. The photocurrent is measured under simulated 100 mW cm⁻² AM 1.5G irradiation using an Oriel 96000 solar simulator calibrated with a standard Si solar cell. The average PCE is obtained using 15 high performance devices under the same conditions.

The EQE values of the encapsulated devices are measured using a lock-in amplifier (SR810, Stanford Research Systems). The devices are illuminated by monochromatic light from a 150 W xenon lamp passing through an optical chopper and a monochromator. The photon flux is determined by a calibrated standard silicon photodiode.

Mobility measurements are performed with the following diode structures: Al/DR3TBDTT:PC₇₁BM/ETL/Al for electron at the J-V curve in the range of 0–7 V. The charge carrier mobility is calculated using the space-charge limited current (SCLC)⁴⁰ model:

$$J = \frac{9\varepsilon_0\varepsilon_r\mu_0 V^2}{8L^3}exp(0.89\beta\sqrt{\frac{V}{L}})$$

in which J is the current density, L is the film thickness of the active layer, μ_0 is the zero-field electron mobility, ε_r is the relative dielectric constant of the transport medium, ε_0 is the permittivity of free space (8.85 × 10⁻¹² F m⁻¹), β is the field activation factor, and $V (= V_{appl} - V_{bi})$ is the internal voltage in the device, in which V_{appl} is the applied voltage to the device and V_{bi} is the built-in voltage caused by the relative work function difference of the two electrodes.

Detailed encapsulation techniques procedure: We took a bit UVcurable EPOXY resin on the device; and then put a cover glass on the glue; finally UV-curable process was followed when the glue covered the devices.

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Solution-processed carbon dots (CDs) as the electron transport layers (ETLs) significantly enhanced device performance (J_{sc} and FF) and lifetime.