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A Degradable Polydopamine Coating Based on Disulfide-Exchange

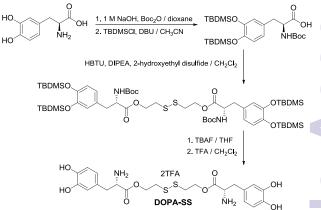
Although the programmed degradation of biocompatible films finds its applications in various fields including biomedic: and bionanotechnological areas, the coating methods have generally been limited to be substrate-specific, not applicable to any kinds of substrates. In this paper, we report a dopamine derivative, which allows for both universal coating various substrates and stimuli-responsive film degradation, inspired by mussel-adhesive proteins. Two dopamine mointies are linked together by the disulfide bond, the cleavage of which enables the programmed film degradation. Mechanisus analysis of the degradable films indicates that the initial cleavage of the disulfide linkage causes a rapid uptake of w molecules, hydrating the films, which leads to rapid degradation. Our substrate-independent coating of degradable films would provide an advanced tool for drug delivery systems, tissue engineering, and anti-fouling strategies

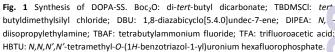
Introduction

Polydopamine-based, substrate-independent coating,¹ inspired by the adhesive property of mussels in nature, has intensively been used in various fields, ranging from biomedicine to energy-related fields.^{2,3} Its extraordinary adhesion property, presumably stemming from the synergistic effects of 1,2-hydroxybenzene (catechol) and amine groups, has led 2-(3,4-dihydroxyphenyl)ethylamine (dopamine) as a minimal building block to coat virtually all substrates, including superhydrophobic surfaces,⁴⁻⁶ electrodes,^{7,8} and even living cell surfaces.^{9,10} Moreover, the derivatization of dopamine also has provided the additional or orthogonal functionalities that expanded the characteristics and applications of the coated surfaces.¹¹⁻¹⁸ For example, the polymerization and cross-linking of 6-nitrodopamine produced photocleavable and self-healing matrices.¹⁴ The polydopamine derivative films that possessed o-methylphenyl aldehyde moieties were used for the attachment of external biomolecules via photo-triggered Diels-Alder reaction.¹⁵ Norepinephrine, having an additional OH group, has been proposed as an ultrasmooth coating unit,¹⁶ and its β -OH group allowed for additional functions, such as a site for ring-opening polymerization.¹⁷ We also have recently reported the use of a perfluorinated dopamine derivative for self-cleaning, superhydrophobic coating on various

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substrates.¹⁸ Although the processability of dopamine has proved multifunctionally useful toward various surfaces, th a formation of polydopamine-based, degradable films remains elusive, because the high adhesiveness is generally noncompatible with degradability.





Stimulus-responsive, degradable coating offers basic bic indispensable tools for drug delivery systems, tissue engineering, cosmetics, and anti-fouling strategies. For example, spatiotemporal release and subsequent delivery of therapeutics to target tissues are achieved by controlle degradation of drug-embedding cargos.¹⁹⁻²¹ The non-specifi adsorption of marine-bio mass onto vessel surfaces also would be eliminated by degradable coating.²² In addition, recer. interest has been focused on degradable cell-surf engineering to manipulate cellular metabolism and activities

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a single-cell level.^{23,24} However, degradable films have mainly been formed by layer-by-layer (LbL) assembly²⁵⁻²⁷ and surfaceinitiated polymerization,^{28,29} which require substrate-specific chemical interactions between a substrate and coating materials. Although it was reported that the LbL procedure could be made substrate-independent with pre-coating of catechol derivatives,^{30,31} it is still highly demanding to develop a method for the universal and degradable coating. In this work, we designed a dopamine derivative that formed stimulus-responsive, degradable films on virtually any substrates.

Results and discussion

We used 3,4-dihydroxy-L-phenylalanine (L-DOPA) as a starting material in our synthetic design, because the dopamine moiety kept preserved with derivatization of the carboxyl group (Fig. 1).^{18,32} After protection of the hydroxyl and amine groups in L-DOPA, two dopamine structures were linked together by the disulfide (S-S) bond, which would be broken to two thiol (SH) groups under reducing conditions. The subsequent deprotection reactions generated a water-soluble dopamine derivative, denoted as DOPA-SS in this paper, and the coating was performed under slightly basic conditions (pH 8.5) at room temperature.

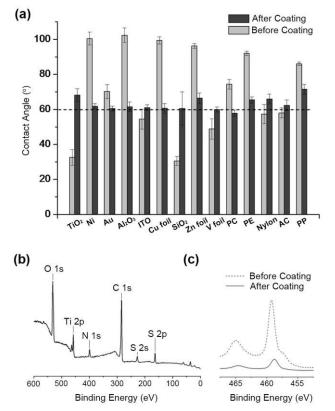
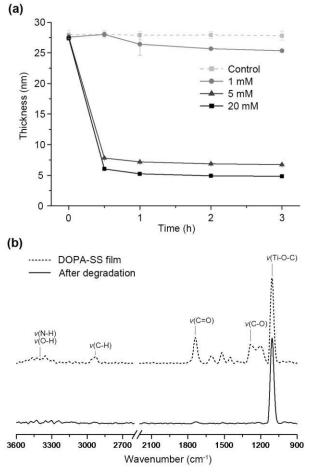
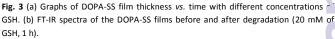


Fig. 2 Characterizations of DOPA-SS films. (a) Static water contact angles of the substrates before (gray) and after (black) DOPA-SS film formation. (b) Wide-scan XPS spectrum of DOPA-SS-coated TiO₂. (c) The Ti(2p) region of the XPS spectra of bare and DOPA-SS-coated TiO₂.

We coated various surfaces, such as titanium oxide (TiO_2) , nickel (Ni), gold (Au), aluminum oxide (Al_2O_3) , indium tin oxide

(ITO), copper foil (Cu foil), silicon oxide (SiO₂), zinc foil (Zn foil), vanadium foil (V foil), polycarbonate (PC), polyethylene (PE), nylon, acryl plate (AC), and polypropylene (PP). The coat proved substrate-independent: after coating the water contact angles of all the samples tested converged to about 60° regardless of the initial values that were significantly variable depending upon the identity of the substrates. (Fig. 2a). The ray photoelectron spectroscopy (XPS) analysis also confirmed the successful coating. For example, the XPS spectrum of the TiO₂ substrate showed the characteristic peaks of DOPA-SS at 399.6 (N 1s), 227.8 (S 2s), and 164.9 eV (S 2p) (Fig. 2b, see ES, Fig. S1⁺ for the XPS spectra of the other substrates). The C(1s) XPS peak was deconvoluted to obtain further information on the coated film: the peaks appearing at 285.9 and 289.0 ev corresponded to the C-X (X = O, N, or S) and the C=O bon(, respectively, confirming the existence of the functional group of DOPA-SS (see ESI, Fig. S2⁺). In contrast, the signal intensitie of TiO₂ (Ti 2p_{3/2}, 458.7 eV; 2p_{1/2}, 464.7 eV) decrease. significantly after coating, indicating that the DOPA-SS (attenuated the characteristic signals of the underlying TiO₂ substrate (Fig. 2c). The ellipsometric thickness of the DOPA film was measured to be 27.5 nm for TiO₂.



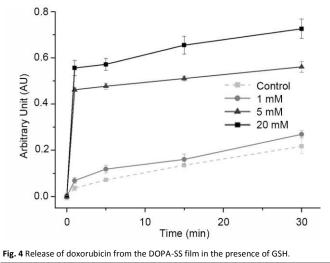


The degradation characteristics of the DOPA-SS film were investigated with glutathione (GSH) as a reducing agent,

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intracellular antioxidant that prevents cell damages by reducing reactive oxygen species.³³ We chose the GSH concentrations of 1, 5, and 20 mM, because the substantial difference in the GSH concentration between inner (~10 mM) and exterior (~2 μ M) cells in this range³⁴ has been utilized for the controlled release of drug in the target cells.^{35,36} The DOPA-SS film degradation, monitored by the ellipsometric measurements, was found to be concentration-dependent (Fig. 3a). The 1 mM of GSH had a negligible effect on degradation compared with a control (Tris, pH 7.4), but higher concentrations of GSH (5 mM and 20 mM) led to fast film degradation. The XPS peaks from TiO₂ (Ti $2p_{3/2}$, 458.7 eV; $2p_{1/2}$, 464.7 eV) were recovered after GSH treatment, which additionally supported the removal of the DOPA-SS film from the TiO₂ substrate (see ESI, Fig. S3⁺). The Fourier-transform infrared (FT-IR) spectra showed the chemical states of the DOPA-SS films before and after degradation (Fig. 3b). The peaks from DOPA-SS at 3500-3300 (v(N-H) and v(O-H)), 2930 (v(C-H)), 1739 (v(C=O)), and 1282 and 1204 cm⁻¹ (v(C-O)) were barely observed after treatment of GSH (20 mM; 1 h), while the peak at 1107 cm⁻¹ (v(Ti-O-C))³⁷ remained unchanged. In addition, the characteristic peaks of the indole structure³⁸ at 1606 (v_{ring}(C=C)), 1520 (v_{ring}(C=N)), and 1606 cm⁻¹ (v_{ring}(CNC)) were not observed after degradation. The IR results indicated that the film was removed except for the adlayer that was covalently linked with TiO₂.



As a proof-of-concept demonstration for controlled drug release systems, we co-deposited a therapeutic, doxorubicin, with DOPA-SS, and investigated its release behaviors with GSH as a reducing agent. Fluorescence spectrometry analysis showed the concentration-dependent release profiles of doxorubicin (Fig. 4). When the concentration of GSH was 5 mM or 20 mM, doxorubicin was released mostly in 1 min, and reached a plateau after 30 min. In contrast, 1 mM of GSH and the control (Tris pH 7.4 without GSH) showed a passive release of doxorubicin. Doxorubicin-free DOPA-SS films showed a negligible fluorescence signal, indicating that the degraded components of DOPA-SS films did not interfere with the detection of doxorubicin (data not shown).

To investigate mechanistic details of the film degradation process over time, quartz crystal microbalance with dissipation (QCM-D) experiments were conducted. In Fig. 5, the change in frequency (Δf) and dissipation (ΔD) represented the state of the mass change and film viscoelasticity, respectively, upo-GSH addition to an adsorbed DOPA-SS film.³⁹⁻⁴¹

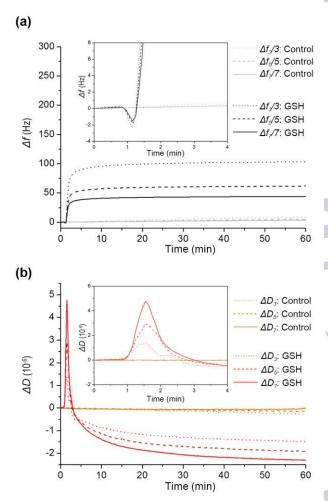


Fig. 5 QCM-D spectra of the DOPA-SS-coated TiO₂. (a) Normalized frequency change $(\Delta f_n/n)$ where *n* is the odd overtone integer and (b) energy dissipation change (ΔD) as functions of time were measured after the injection of GSH (20 mM). Simultaneous measurements were recorded at three overtones (*n*=3, 5, and 7).

Almost immediately, a sharp increase in frequency ($\Delta f > 0$) was observed with the injection of 20 mM GSH, which was indicative of a rapid mass decrease at the surface layer (Fig. 5a). Of interest is the observation that the dissipation increased sharply ($\Delta D > 0$) with a decrease in frequency ($\Delta f < 0$) right after injection of GSH, and it gradually decreased with the increase in frequency (Fig. 5b). Based on the QCM-D data, we believed that the introduction of GSH to the DOPA-SS film s induced the initial decrease in the degree of crosslinking by reducing the S-S bond to the SH groups, which led to increas in hydration mass ($\Delta f < 0$, frequency decrease) and in the corresponding film viscoelasticity ($\Delta D > 0$, energy dissipation increase) at the very early stage of degradation.^{42,43} The relatively high degree of hydration (as indicated by the ΔL , __, ratio) would subsequently increase the accessibility of G^{cv}.

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which in turn enabled the rapid degradation of the DOPA-SS films within a short period of time. Simultaneously, the spread between the three experimentally measured overtones (n=3, 5, and 7) increased dramatically (Fig. 5b), indicating that the physical characteristics of the DOPA-SS films changed from a rigid layer to a soft, viscoelastic layer during the initial stage of the degradation process. In contrast, the DOPA-SS films without GSH (control) showed negligible changes in Δf and ΔD , supporting that the degradation of the DOPA-SS films was caused by GSH, which cleaved the disulfide network. In addition, negligible differences in Δf and ΔD were observed for the 1 mM GSH solution compared with a control, while the 5 mM GSH solution led to lower mass decrease than the 20 mM GSH solution, which was consistent with the ellipsometric analysis (see ESI, Fig. S4†).

Conclusions

In summary, we reported the substrate-independent coating of degradable polydopamine that contained the disulfide linkage. We believe that our work showed several advantages in the area of functional thin films: (1) on-demand degradability with universality: many technologically important applications, including drug delivery systems, frequently require the controlled degradation of materials and films in response to external stimuli. The formation of degradable films is also highly demanded to be universal, not requiring substrate-specificity, exemplified by a metal-organic film of ferric ion and tannic acid.⁴⁴ Our work clearly adds the important property of controlled degradability to the polydopamine films, while preserving their universality of coating; (2) cytocompatibility: the cytocompatible degradability of the DOPA-SS films is highly advantageous for various biomedical applications, such as drug delivery systems, tissue engineering, and cell therapy, where the intimate contact between living cells and coating materials is required. Our system also could be applied to cell surface engineering, where the controlled formation and degradation of films should be performed in a cytocompatible fashion;⁴⁵ (3) design flexibility: it is possible to develop various building blocks of coating for certain applications by incorporating additional functionalities of interest to L-DOPA.

Experimental Section

Film formation and degradation

All substrates were washed thoroughly with ethanol before coating processes. The buffer solution (10 mM Tris, pH 8.5) of DOPA-SS (2.5 mM) was prepared and carefully poured over a substrate for coating. After 3 h, the substrate was washed with DI water and dried under a stream of argon gas. The substrate was coated again with a freshly prepared Tris solution of DOPA-SS for 3 h, and washed with DI water, and dried under a stream of argon gas. The buffer solution (10 mM Tris, pH 7.4) of GSH (1, 5, or 20 mM) was used for DOPA-SS film

degradation. DOPA-SS-coated substrates were immersed in the prepared GSH solution for predetermined time, washed with DI water, and dried under a stream of argon gas.

Release profiles of doxorubicin

The buffer solution (10 mM Tris, pH 8.5) of DOPA-SS (2.5 m M¹) and doxorubicin hydrochloride (125 μ M) was prepared, an i carefully poured over TiO₂ substrates for coating. After 3 h, the TiO₂ substrates were washed with DI water and dried under i stream of argon gas. For GSH-mediated film degradation and release of doxorubicin, the substrates were incubated in i buffer solution (10 mM Tris, pH 7.4) of GSH (1, 5, or 20 mM) for predetermined time. An aliquot was taken from the solution, and its fluorescence intensity was measured by fluorescence spectrometry (λ_{ex} = 480 nm; λ_{em} = 580 nm) for th quantification of released doxorubicin.

QCM-D analysis of DOPA-SS-coated films

Quartz crystal microbalance with dissipation (QCN measurements for frequency change, Δf , and dissipation change, ΔD , were performed with a Q-sense E4 system (Bi Scientific). A titanium-coated crystal was used (Biolin Scientific). Tris buffered solution (10 mM, pH 7.4) was used as a flow (f... rate: 50 µL/min). The DOPA-SS-coated crystal was mounted a room temperature and prewashed with a buffer solution for 20 min. The degradation of the DOPA-SS films was performe by injecting the buffer solution that contained GSH (20 mM and time-course changes in the Δf and ΔD were measured simultaneously at the seven different overtones for 1 h. The crystal was excited at its fundamental frequency (5 MHz), and the measurements were performed at the first, third, fiftl , seventh, ninth, eleventh, and thirteenth overtones. corresponding to 5, 15, 25, 35, 45, 55, and 65 MH respectively. For the analysis, Δf and ΔD of the third, fifth, and seventh overtone of a QCM crystal were used.

Characterizations

All synthesized compounds were characterized by nuclear magnetic resonance spectroscopy (NMR, Inova) operated at the ultrashield of 400 MHz. High resolution mass spectrometry (HR-MS, Bruker Daltonik) was used for the characterization c DOPA-SS. The X-ray photoelectron spectroscopy (XPS) st was performed with a VG-Scientific spectrometer (mode' Sigma Prove) with a monochromatized Al K α X-ray source (1486.6 eV). Emitted photoelectrons were detected by a mult^{*} channel detector at a take-off angle of 90° relative to the surface. During the measurements, the base pressure was 8. × 10⁻⁸ Torr. Survey spectra were obtained at a resolution of eV from 2 scans, and high-resolution spectra were acquired a' a resolution of 0.05 eV from 15 scans. Contact angle measurements were performed with Phoenix 300 apparatus (Surface Electro Optics Co.) equipped with a video camera. Th. static contact angles of 2-µL water droplets were measured at more than five different locations on each sample, and the average values were reported in this paper. The fluorescence intensities of doxorubicin (λ_{ex} = 480 nm; λ_{em} = 580 nm) were measured with a Varioskan Flash Multimode Reader (The

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Scientific). The thickness of the DOPA-SS films was measured with a Gaertner L116s ellipsometer (Gaertner Scientific Corporation) equipped with a He-Ne laser (632.8 nm) at a 70° angle of incidence. A refractive index of 1.46 was used for all the films. FT-IR spectra were obtained in vacuum with an IFS-66v/S FTIR spectrometer (Bruker).

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