## **RSC Advances**



This is an *Accepted Manuscript*, which has been through the Royal Society of Chemistry peer review process and has been accepted for publication.

Accepted Manuscripts are published online shortly after acceptance, before technical editing, formatting and proof reading. Using this free service, authors can make their results available to the community, in citable form, before we publish the edited article. This Accepted Manuscript will be replaced by the edited, formatted and paginated article as soon as this is available.

You can find more information about *Accepted Manuscripts* in the **Information for Authors**.

Please note that technical editing may introduce minor changes to the text and/or graphics, which may alter content. The journal's standard <u>Terms & Conditions</u> and the <u>Ethical guidelines</u> still apply. In no event shall the Royal Society of Chemistry be held responsible for any errors or omissions in this *Accepted Manuscript* or any consequences arising from the use of any information it contains.



www.rsc.org/advances

Cite this: DOI: 10.1039/c0xx00000x

www.rsc.org/xxxxx

## ARTICLE TYPE

## A Bifunctional Oxygen Electrocatalyst from Monodisperse MnCo<sub>2</sub>O<sub>4</sub> Nanoparticles to Nitrogen enriched Carbon Nanofibers

Chaohe Xu,<sup>*a,b*</sup> Meihua Lu,<sup>*a*</sup> Yi Zhan,<sup>*a*</sup> and Jim Yang Lee<sup>*a*</sup>

Received (in XXX, XXX) XthXXXXXXX 20XX, Accepted Xth XXXXXXXX 20XX 5 DOI: 10.1039/b000000x

Monodisperse  $MnCo_2O_4$  nanoparticles supported on nitrogen enriched carbon nanofibers (NCF) exhibited synergy of component interaction that resulted in higher ORR and OER activities than commercial Pt/C catalysts.

- <sup>10</sup> The higher theoretical energy densities of metal-air batteries are their most noticeable advantage over lithium-ion batteries for electric vehicle propulsion applications. For rechargebility the oxygen reduction reaction (ORR) and the oxygen evolution reaction (OER) in the cathode must have high electrochemical
- <sup>15</sup> reversibility. These reactions are however kinetically challenged necessitating the presence of bifunctional oxygen electrocatalysts equally adept at ORR and OER<sup>1</sup>. Although noble metals have shown excellent electrocatalytic activities (e.g. Pt for ORR and RuO<sub>2</sub> or IrO<sub>2</sub> for OER) in aqueous solution, their scarcity and <sup>20</sup> high cost are deployment challenges.<sup>2, 3</sup> The interest to develop

low cost alternatives to the noble metals is clear and vital.

Mixed-valence transition metal oxides with a spinel structure have been used as bifunctional oxygen electrocatalysts in alkaline solutions. Among them those based on cobalt oxides have drawn

- <sup>25</sup> the most potential because of reasonably good electrocatalytic activity, low cost, ease of preparation and good chemical stability. The ORR and OER electrocatalytic activities could be increased further by partially substituting the cobalt ions with Ni and Mn ions<sup>4</sup>. Despite these improvements, the ternary oxides
- $_{30}$  still could not match the performance of Pt-based catalysts. For example, at an ORR current density of ~3 mA/cm² (the current density measured at the half-wave potential of 20 wt% Pt/C at 5 mV/s), the oxygen reduction potentials were 0.49, 0.59 and 0.68 V (vs RHE) for Co<sub>3</sub>O<sub>4</sub>, Co<sub>2</sub>MnO<sub>4</sub> and a Co<sub>3</sub>O<sub>4</sub>/Co<sub>2</sub>MnO<sub>4</sub>
- <sup>35</sup> composite respectively, significantly more negative than the 0.80 V for Pt/C. The limiting current densities of oxygen reduction on these oxides were also lower than that of  $Pt/C^5$ . In a different study, while Pt nanoparticles on carbon nanotubes could deliver 125 mA/cm<sup>2</sup> at a loading of only 0.1 mg/cm<sup>2</sup>, MnCo<sub>2</sub>O<sub>4</sub> had to be
- <sup>40</sup> loaded to 14 mg/cm<sup>2</sup> on carbon black to yield a current density of 300 mA/cm<sup>2</sup>.<sup>6</sup> Clearly the mass catalytic activity of oxide-based electrocatalysts leaves plenty of room for improvement. Recent reports have shown that hybrid structures of oxides and carbon nanofoams could improve the application performance through
- <sup>45</sup> synergistic interactions if there is strong coupling between the components. For instance, the Dai group covalently bonded Co<sub>3</sub>O<sub>4</sub> and MnCo<sub>2</sub>O<sub>4</sub> nanoparticles to graphene and reported oxygen electrocatalytic activities higher than those of Co<sub>3</sub>O<sub>4</sub>,

 $MnCo_2O_4$  or graphene alone. The activities were comparable to <sup>50</sup> that of Pt/C and more stable than the later in KOH solution<sup>7, 8</sup>.

In a separate development, hetero-atom doped nanocarbons were also found to be highly ORR active. For instance, nitrogendoped carbon nanotube arrays have shown higher electrocatalytic activity and greater chemical stability than Pt on carbon for ORR 55 under alkaline conditions<sup>9</sup>. Pt supported on nitrogen-doped carbon also surpassed Pt on pristine carbon in ORR performance <sup>10</sup>. Research over the years has ratified the ORR activity of nitrogen-doped carbon where the N-C-N bonding motifs are the most likely active sites<sup>11-13</sup>. In particular nitrogen doping of 60 graphitic carbon lattices (graphitic-N) forms one of the most active ORR catalytic sites by facilitating the electron transfer from the carbon conduction band to the anti-bonding orbitals of oxygen<sup>14, 15</sup>. However, graphitic-N is not as easily formed as pyridinic and pyrrolic-N by direct nitrogen doping because of a 65 higher energy of formation. A better strategy is to start with precursors which can introduce more graphitic-N in the nanocarbon; for which nitrogen-containing conducting polymers are good choices<sup>16</sup>. For example, polypyrrole (PPy) has been used to produce microporous nanocarbon with graphitic-N to <sup>70</sup> pyridinic-N ratios as high as 5.89<sup>17</sup>.

Based on these background information we have developed a MnCo<sub>2</sub>O<sub>4</sub> nanoparticle-nitrogen enriched carbon nanofiber (NCF) composite with an ORR performance as good as the commercial Pt/C catalyst; and an OER performance which is even better. The 75 composite was prepared by a solvothermal method (See Experimental details in ESI), using nitrogen enriched carbon nanofibers which were derived from pyrolysis of PPy nanowires (40~60 nm in diameters, see Figure S1) in nitrogen at 800 °C for 2 h. In a typical preparation, Mn and Co precursors in a ratio of 80 Mn/Co=1:2, ammonia solution; and H<sub>2</sub>O were added to a NCF suspension in ethanol. After ageing at 80 °C for several hours, the suspension was solvothermally treated at 150 °C and the final product was recovered by centrifugation. As shown in the typical TEM and SEM images of Figure 1, there was uniform deposition 85 of 2-4 nm nanoparticles on the surface of the NCFs, which were about 40~60 nm in diameter and several µm in length (Figure S2). All the peaks in the X-ray diffraction (XRD) patterns of the composite indexed well with the MnCo<sub>2</sub>O<sub>4</sub> spinel structure (Figure S3). The HRTEM image in the inset of Figure 1b also 90 shows that the MnCo<sub>2</sub>O<sub>4</sub> nanoparticles were highly crystalline. The MnCo<sub>2</sub>O<sub>4</sub> loading in the NCFs was ~35 wt. % according to thermal analysis (Figure S4). Scanning transmission electron

microscopy (STEM) analysis of elemental distribution in the composite (Figure 2) measured homogeneity of Mn, Co, N and O distributions throughout the composite. Hence it may be concluded that the  $MnCo_2O_4$  nanoparticles had uniformly <sup>5</sup> decorated the NCF.



**Figure 1**. (a-b) TEM and (c-d) SEM images of the MnCo<sub>2</sub>O<sub>4</sub>-<sup>35</sup> NCF composite; (e) XPS spectrum of NCFs (red), MnCo<sub>2</sub>O<sub>4</sub>/NCF composite (black), and pure MnCo<sub>2</sub>O<sub>4</sub> nanoparticles (blue); (f) N1s spectrum of NCFs.

X-ray photoelectron spectroscopy (XPS) of the NCFs and their 40 composite (Figure 1e-f) confirmed the presence of N in very high amounts (~11.5 at.%). The N1s spectrum of NCF could be deconvoluted into two components (Figure 1f) representing pyridinic-N (397.8 eV) and graphitic-N (400.4 eV) respectively. The graphitic-N content calculated as such was as high as 54.6%. 45 Such a high graphitic-N content, when compounded with MnCo<sub>2</sub>O<sub>4</sub> nanoparticles which have previously shown bifunctional electrocatalytic activities for oxygen<sup>18</sup>, may result in a greater oxygen electrocatalysis performance through improvements in the charge transfer process and in electron 50 conduction through the catalyst system. The Co/Mn ratio of  $\sim$ 2 as measured by XPS was in good agreement with the starting precursor ratio. High resolution XPS also revealed the interaction between MnCo2O4 nanoparticles and NCF. Relative to pure MnCo<sub>2</sub>O<sub>4</sub>, there was a general decrease in the O1s, Mn2p<sub>3/2</sub> and 55 Co2p<sub>3/2</sub> binding energies for the MnCo<sub>2</sub>O<sub>4</sub>/NCF composite

(Figure S5). The pyridinic-N1s (398.1 eV) and graphitic-N1s (400.7 eV) peaks of the composite were  $\sim 0.3$  eV higher than those in NCF (Figure S6a-b); while the C1s peak of the

composite was shifted to slightly lower binding energy (Figure 60 S7). Collectively these are indications of electron migrations from the NCF to metal oxide nanoparticles; and from the nitrogen lone pair to the sp<sup>2</sup> carbon skeleton<sup>7</sup>. There was therefore strong electronic interaction between MnCo<sub>2</sub>O<sub>4</sub> and NCF in the composite. Such interaction could render the oxide nanoparticles <sup>65</sup> more conducting and more electrochemically active than a binary mixture of NCF and MnCo<sub>2</sub>O<sub>4</sub> nanoparticles.

For the MnCo<sub>2</sub>O<sub>4</sub>/NCF composite, Figure S6c-d also show an asymmetric O1s peak which could be deconvoluted into two distinguishable peaks at 529.9 and 531.1 eV (530.2 and 531.5 eV <sup>70</sup> respectively for MnCo<sub>2</sub>O<sub>4</sub>) which correspond well with lattice oxide oxygen and surface adsorbed oxygen-containing species (possibly hydroxide or water)<sup>19</sup>. The higher relative intensity and integrated peak area of adsorbed oxygen species on the composite may be used to suggest a stronger oxygen affinity in the former, <sup>75</sup> which predisposes the composite to oxygen electrocatalysis.



Figure 2. Elemental mapping images of C, N, Mn, Co and O.

The ORR performance of MnCo<sub>2</sub>O<sub>4</sub>-NCF composite was evaluated by cyclic voltammetry on a glassy carbon electrode in 95 O<sub>2</sub>- or N<sub>2</sub> saturated 0.1 M KOH solutions. Unsupported MnCo<sub>2</sub>O<sub>4</sub> nanoparticles (with comparable sizes of 3~5 nm, Figure S8) and NCFs were also evaluated under the same conditions for comparison. Figure 3a shows the cyclic voltammograms (CVs) of various samples together with a standard commercial Pt/C 100 catalyst (40 wt.% Pt on graphitized carbon, Aldrich). The MnCo<sub>2</sub>O<sub>4</sub>-NCF composite exhibited a more positive ORR onset potential (-0.08 V vs. Ag/AgCl) and a higher cathodic peak current than those of MnCo2O4 nanoparticles and NCFs, suggesting a synergy of functions after the component integration. <sup>105</sup> The composite was more similar to Pt/C where the onset potential was about -0.06 V. The ORR kinetics was also measured by a rotating-disk electrode in O2-saturated 0.1 M KOH solution. Figure 3b shows the typical linear sweep voltammograms (LSVs) of MnCo<sub>2</sub>O<sub>4</sub>-NCF composite. The Koutecky-Levich plots were 110 all linear and parallel to one another, confirming first-order reaction kinetics. The electron transfer numbers (n) calculated from the slopes in the limited current region (-0.4 to -0.6 V)Figure 3c) were  $\sim$ 3.96. These measurements indicate that MnCo<sub>2</sub>O<sub>4</sub> nanoparticles supported on NCFs are effective for a 4-115 electron ORR. By comparison the electron transfer numbers of NCFs, VC-72 carbon, and physical mixtures of MnCo<sub>2</sub>O<sub>4</sub> with

NCFs; and of  $MnCo_2O_4$  with VC-72 carbon; were 2.45, 2.16, 3.64, and 2.30 respectively (Figure S9). The electron transfer number was higher in the NCF mixture than in the VC-mixture because the higher intrinsic ORR activity of the NCFs provided a better

- <sup>5</sup> match with the intrinsic activity of MnCo<sub>2</sub>O<sub>4</sub>. The LSVs of the composite, various control samples, and of Pt/C, are compared in Figure 2d. The current density of the composite was higher than that of Pt/C in the mixed-controlled and diffusion-controlled potential regions. The half-wave potential of MnCo<sub>2</sub>O<sub>4</sub>-NCF was
- <sup>10</sup> also slightly more positive than Pt/C (-0.21 V compared with -0.23 V for Pt/C). More importantly the MnCo<sub>2</sub>O<sub>4</sub>-NCF composite was very stable in 0.1 M KOH, where the 17% decrease of ORR activity after 30000 s of continuous operation at -0.4V compares favourably with the 34% decrease on Pt/C (Figure 4). Hence the
- <sup>15</sup> MnCo<sub>2</sub>O<sub>4</sub>-NCF composite surpasses Pt/C in both the activity and stability of ORR.



Figure 3. Electrochemical performance of the MnCo<sub>2</sub>O<sub>4</sub>-NCF composite as an ORR catalyst. (a) CV of MnCo<sub>2</sub>O<sub>4</sub>, NCF, 35 MnCo<sub>2</sub>O<sub>4</sub>-NCF, and Pt/C on GC electrodes in O<sub>2</sub>-saturated or N<sub>2</sub>-saturated 0.1 M KOH. (b) LSV of MnCo<sub>2</sub>O<sub>4</sub>-NCF composite at 5 mV/s. (c) The corresponding Koutecky-Levich plots of MnCo<sub>2</sub>O<sub>4</sub>-NCF composite at different potentials. (d) LSV of MnCo<sub>2</sub>O<sub>4</sub>-NCF and various controls at 5 mV/s and a rotation 40 speed of 900 rpm.

The OER performance of the MnCo<sub>2</sub>O<sub>4</sub>-NCF composite was also measured, by extending the anodic potential limit to 1.0 V (Figure 5a). In a 0.1 M KOH solution, OER occurred earlier (at 45 less positive potentials) on the MnCo<sub>2</sub>O<sub>4</sub>-NCF and current densities were also generally higher than those of Pt/C and NCFs. The onset potential of OER on MnCo<sub>2</sub>O<sub>4</sub>-NCF was about 0.6 V. While the OER catalytic properties of MnCo<sub>2</sub>O<sub>4</sub>-NCF are superior to Pt/C, they are still below the performance of precious 50 metal oxides (RuO<sub>2</sub> or IrO<sub>2</sub>) which are the best OER catalysts for commercial electrolysis<sup>20, 21</sup>. In order to evaluate the overall suitability of MnCo<sub>2</sub>O<sub>4</sub>-NCF as a bifunctional oxygen electrocatalyst, we used the difference between the half-wave potential of ORR on Pt/C and the potential for water electrolysis

<sup>55</sup> at 10 mA/cm<sup>2</sup> as an indicator. The smaller the difference, the closer is the catalyst to being truly bifunctional. The value was 1.04V for the MnCo<sub>2</sub>O<sub>4</sub>-NCF composite; which is smaller than previous results<sup>5</sup>. The difference cannot be measured for Pt/C

(black curves) and NCFs (red curves) with OER potentials lower 60 than 1.0 V (Figure 5a). The composite stability under the OER condition was evaluated by CV. Figure 5b shows that the current density decreased to ~15 mA/cm<sup>2</sup> after 400 cycles. The retention of only 83.3% of the initial current density indicates that improvements are needed to increase the OER stability of this 65 material.



**Figure 4.** Chronoamperograms of MnCo<sub>2</sub>O<sub>4</sub>-NCF and Pt/C on <sup>80</sup> GC electrodes at -0.4 V (vs. Ag/AgCl) in O<sub>2</sub>-saturated 0.1 M KOH solution.



**Figure 5.** (a) Suitability of  $MnCo_2O_4$ -NCF as a bifunctional catalyst for ORR and OER in comparison with NCF and Pt/C. (b) CV curves recorded at 1<sup>st</sup> cycle, 100<sup>th</sup> cycle, and 400<sup>th</sup> cycle for MnCo<sub>2</sub>O<sub>4</sub>-NCF composites. Rotation speed = 900 rpm.

In summary, we have prepared a MnCo<sub>2</sub>O<sub>4</sub> nanoparticlenitrogen enriched carbon nanofiber composite with an ORR performance as good as the standard commercial Pt/C catalyst and an even better OER performance. The improved <sup>100</sup> electrochemical performance correlated well with the abundance of ORR-active graphitic-N sites in NCFs, and the effective integration of MnCo<sub>2</sub>O<sub>4</sub> nanoparticles and nitrogen enriched carbon nanofibers to synergize charge transfer and post-charge transfer electron conduction. The use of conducting polymer-<sup>105</sup> derived carbon nanostructures with a high nitrogen content as an active support for non-precious metal catalysts should be explored further for the preparation of low-cost bifunctional oxygen electrocatalysts.

<sup>110</sup> This research is financially supported by the research grant (R-265-000-436-305) of Advanced Energy Storage Programme from the SERC, Singapore.

## Notes and references

<sup>a</sup> Department of Chemical and Biomolecular Engineering, National 115 University of Singapore, 10 Kent Ridge Crescent, Singapore 119260,

**RSC Advances Accepted Manuscript** 

Singapore; Fax: +65-67791936; Tel: +65-65162899; Email: <u>cheleejy@nus.edu.sg</u>;

<sup>b</sup> Department of Mechanical Engineering, National University of Singapore, 9 Engineering Drive 1, Singapore 117576, Singapore.

- 5 † Electronic Supplementary Information (ESI) available: [Aditional TEM, XDR, XPS and LSV curves]. See DOI: 10.1039/b000000x/
- 1. F. Y. Cheng and J. Chen, *Chemical Society Reviews*, 2012, **41**, 2172-2192.
- 10 2. S. Guo, D. Li, H. Zhu, S. Zhang, N. M. Markovic, V. R. Stamenkovic and S. Sun, *Angewandte Chemie International Edition*, 2013, **52**, 3465-3468.
- 3. S. J. Guo and S. H. Sun, *Journal of the American Chemical Society*, 2012, **134**, 2492-2495.
- Y. Liang, Y. Li, H. Wang and H. Dai, *J Am Chem Soc*, 2013, 135, 2013-2036.
- D. Wang, X. Chen, D. G. Evans and W. Yang, *Nanoscale*, 2013, 5, 5312-5315.
- 6. F. Bidault, D. J. L. Brett, P. H. Middleton and N. P. Brandon,
- 20 *Journal of Power Sources*, 2009, **187**, 39-48.
- Y. Y. Liang, H. L. Wang, J. G. Zhou, Y. G. Li, J. Wang, T. Regier and H. J. Dai, *Journal of the American Chemical Society*, 2012, 134, 3517-3523.
- Y. Liang, Y. Li, H. Wang, J. Zhou, J. Wang, T. Regier and H.
  Dai, *Nat Mater*, 2011, **10**, 780-786.
- K. Gong, F. Du, Z. Xia, M. Durstock and L. Dai, Science, 2009, 323, 760-764.
- F. B. Su, Z. Q. Tian, C. K. Poh, Z. Wang, S. H. Lim, Z. L. Liu and J. Y. Lin, *Chemistry of Materials*, 2010, **22**, 832-839.
- 30 11. T. P. Fellinger, F. Hasche, P. Strasser and M. Antonietti, Journal of the American Chemical Society, 2012, 134, 4072-4075.
- J. D. Wiggins-Camacho and K. J. Stevenson, Journal of Physical Chemistry C, 2011, 115, 20002-20010.
- 35 13. H. Kim, K. Lee, S. I. Woo and Y. Jung, *Physical Chemistry Chemical Physics*, 2011, **13**, 17505-17510.
- J. Xu, G. Dong, C. Jin, M. Huang and L. Guan, *ChemSusChem*, 2013, 6, 493-499.
- 15. P. Wang, Z. Wang, L. Jia and Z. Xiao, *Physical Chemistry* 40 *Chemical Physics*, 2009, **11**, 2730-2740.
- G. Wu, N. H. Mack, W. Gao, S. G. Ma, R. Q. Zhong, J. T. Han, J. K. Baldwin and P. Zelenay, *Acs Nano*, 2012, 6, 9764-9776.
- 17. F. Su, C. K. Poh, J. S. Chen, G. Xu, D. Wang, Q. Li, J. Lin and 45 X. W. Lou, *Energy & Environmental Science*, 2011, **4**, 717-
- As X. W. Lou, Energy & Environmental Science, 2011, 4, 71-724.
- H. L. Wang, Y. Yang, Y. Y. Liang, G. Y. Zheng, Y. G. Li, Y. Cui and H. J. Dai, *Energy & Environmental Science*, 2012, 5, 7931-7935.
- 50 19. F. Y. Cheng, J. A. Shen, B. Peng, Y. D. Pan, Z. L. Tao and J. Chen, *Nature Chemistry*, 2011, **3**, 79-84.
- T. Reier, M. Oezaslan and P. Strasser, *Acs Catalysis*, 2012, 2, 1765-1772.
- 21. W. Hu, Y. Q. Wang, X. H. Hu, Y. Q. Zhou and S. L. Chen, 55 *Journal of Materials Chemistry*, 2012, **22**, 6010-6016.

Page 5 of 5