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Herringbone to Cofacial Solid State Packing via H-bonding in Diketopyrrolopyrrole (DPP) based Molecular Crystals: Influence on Charge Transport

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The mono alkylation of DPP derivatives leads to cofacial π-π stacking via H-bonding unlike their di-alkylated counterpart, which exhibits classical herringbone packing pattern. Single crystal organic field-effect transistor (OFET) measurements reveal significant enhancement of charge carrier mobility for mono-hexyl DPP.

The performance of optoelectronic devices based on organic semiconductors is limited by poor charge transport. The charge carrier mobility (\(\mu\)) in molecular semiconductors is strongly influenced by a variety of factors such as purity, traps, defects and molecular stability.1 To surmount these limitations, molecular engineering has taken a centre stage.2-4 Large numbers of new organic functional materials have been synthesized to achieve the best performance for devices such as organic photovoltaics (OPVs) and organic field effect transistors (OFETs).5-7 In this regard, diketopyrrolopyrrole (DPP) based \(\pi\)-conjugated systems have shown promise as a versatile material with excellent charge transport properties.8, 9 Yun-Hi Kim and co-workers have reported one of the highest p-type charge carrier mobilities for DPP-based polymer, surpassing the mobility of amorphous silicon.10 Our group has also observed one of the best electron mobility for DPP based \(\pi\)-conjugated low-band gap polymer.11 Inspired by these results, we envisioned to investigate the structure-property correlation of DPP monomers to explore the full potential as a molecular material for organic electronics.

The DPP-based monomers were synthesized by pseudo-Stobbe condensation reaction followed by di-alkylation of the amide groups present in the DPP chromophore.12 While following this procedure a minor product was also separated along with the expected di-alkylated DPP derivatives. The highly crystalline minor product was identified as mono-alkylated DPP by \(^1\)H, \(^{13}\)C and ESI mass spectroscopy (Electronic Supplementary Information, ESI†). To our surprise, single crystal structure determination reveals classical herringbone12 and cofacial layered structure for di-and mono-hexyl DPP, respectively. This encouraged us to enhance the yield of mono-hexyl DPP by controlling the reaction condition, shown in Scheme 1 and pursue further with an aim to study the comparative charge transport properties of this important class of material which was ignored previously. Herein, we report the synthesis and characterization of mono-alkylated DPP for the first-time. Detailed studies have unveiled several unusual features including the role of H-bonding to influence the favourable packing for enhanced charge transport in single crystals of mono-hexyl DPP than its di-alkylated counterpart.

Scheme. 1 In top, synthesis of PDPP-MH and TDPP-MH starting from their corresponding DPP molecules has been shown. The chemical structures of four different DPP molecules are shown at the bottom.

The optical properties of mono-hexyl DPP molecules were almost similar to their di-alkylated analogues with the appearance of the dual band structure in absorption and emission spectra (Figure 1 (a) and (b)). The low energy band in the absorption spectrum showed vibronic splitting for both mono-hexyl derivatives. But there was minor blue shift (10 nm) of the lowest energy band for TDPP-MH (\(\lambda_{\text{max}}\) at 500, 540 nm) as compared to TDPP-DH (\(\lambda_{\text{max}}\) at 510, 550 nm) indicating increase in band gap while optical band gap remained almost same for both the phenyl coupled DPP (PDPP) derivatives. Both PDPP-MH and TDPP-MH showed blue-shifted vibronic emissive band in comparison with the di-hexyl DPP molecules. The hypsochromic shift was measured to be \(\sim10\) nm for TDPP-MH while it was 25 nm for PDPP-MH. As a result, we observe a significant decrease in Stokes shift for PDPP-MH (0.19 eV) as compared to PDPP-DH (0.32 eV), but for TDPP-MH and TDPP-DH the Stokes shift was almost similar (0.4-0.5 eV). The reduction in Stokes shift suggests a decrease in torsion in PDPP-MH. Further, density functional theory (DFT)
calculations have corroborated our assumption. The detailed optical properties of mono-hexyl derivatives are given in Table S1 in the ESI†. Theoretical studies have shown that in the absence of alkyl chain the repulsive interaction between ortho-
hydrogen of phenyl group and a nearby hydrogen atom attached to the alkyl group vanishes and the phenyl group appears almost in the same plane with DPP core with torsion of 8° enhancing electronic coupling between donor and acceptor moieties (Figure 1d). On the other hand, the other phenyl ring remains as twisted as in PDPP-DH by 36° due to the above said repulsive interaction. Calculations on TDPP-MH predicts the trans-
orientation of two thiophene units with respect to each other and both the donor groups lie in the same plane with the acceptor having a minimum torsion of less than 1°.

The consequence of higher planarity in TDPP-MH is the stronger intramolecular electronic coupling between donor and acceptor moieties resulting in higher molar extinction coefficient for TDPP-MH (20300 L mol⁻¹ cm⁻¹) than the PDPP-MH (15000 L mol⁻¹ cm⁻¹) as measured experimentally. Fluorescence quantum yield decreases from 80% in PDPP-MH to 72% in TDPP-MH which is attributed to the heavy atom effect facilitating the intersystem crossing rates. On the similar line, fluorescence excited state lifetime also decreases from PDPP-MH to TDPP-MH as shown in Figure 1C. The frontier molecular orbital energies were measured from cyclic voltammetry (CV) experiment. The HOMO and LUMO energies are summarized in Table S1 (Electronic Supplementary Information, ESI†). The HOMO energy level destabilized by 0.1 eV in TDPP-MH from PDPP-MH while LUMO stabilized by ~ 0.2 eV due to the change of phenyl with thiophene in the donor moiety.

The single crystal X-ray diffraction study reveals the molecular arrangements and their interactions in the solid state. Both the molecules crystallize in the monoclinic space group P2₁/n z=4. The unique difference between di- and mono- hexyl DPP derivatives lies in their crystal packing arrangements. The free amide group of mono-alkylated DPP molecules plays a pivotal role to turn around the molecular packing in the unit cell. PDPP-
MH exhibits cofacial layered structure while TDPP-MH demonstrates cofacial herringbone packing due to the intermolecular H-bonding between two amide functionalities in DPP chromophore locking the orientation of the molecules. Thus, intermolecular H-bonding influences the packing pattern of DPP-
MH deviating from their di-hexyl counterparts which show exclusively classical edge-to-face herringbone packing. Both, PDPP-MH and TDPP-MH showed slip-stacked packing arrangement with donor thiophene/phenyl group appears top of the acceptor DPP chromophore, which is a common feature of D-
A-D molecular crystals presumably due to the charge transfer interaction. The shortest π-π packing distances were measured to be 3.48 Å and 3.71 Å for PDPP-MH and TDPP-MH, respectively. Similar to our theoretical findings, we see the variation in degree of torsion between the acceptor moiety and two donor groups in both the molecules due to relief from steric repulsion in the absence of an alkyl chain. Torsion varies from 2° to as high as 24° in PDPP-MH while they were 1° and 10° in TDPP-MH. The N-H-O H-bond distances were calculated as 2.04 Å and 1.94 Å in PDPP-MH and TDPP-MH, respectively. Weak C-H-O interaction was also found to operate in both the crystals with the distances of 2.43 Å and 2.27 Å in PDPP-MH and TDPP-MH.

The transmission electron microscopy (TEM) study shows that the nano-to micron sized structures deposited on copper grid are
single crystalline in nature (Figure 3). The mono-hexyl DPP derivatives form fiber like single crystal while di-hexyl analogues form small plate like crystallites. We believe that the presence of H-bonding interaction is responsible for variation in morphologies for mono-hexyl DPP derivatives. The ordered bright diffraction spots in selected area electron diffraction (SAED) pattern confirms the single crystallinity of microstructures. The bright field TEM images and corresponding SAED patterns from single crystal di-hexyl derivatives have been provided in the ESI†.

The H-bonding interaction was further supported by differential scanning calorimetry (DSC). For TDPP-MH, we have observed the appearance of two endo-and exothermic transitions during heating and cooling cycles, respectively (Figure 4). The first endothermic peak was positioned at 207 °C with an enthalpy change of ~2.3 kJ/mole which matches quite well with the H-bonding energy. This compound melted around 250 °C. The change in enthalpy during melting transition was measured to be 21.5 kJ/mole. But dual peaks were not observed for PDPPEMH. In both heating and cooling cycles only one peak was observed, though the enthalpy change during melting process around 250 °C was higher i.e. ~26.8 kJ/mole. The phase transitions observed in DSC measurements were further confirmed by optical polarising microscopy. For TDPP-MH crystals, we have observed a thermal transition around 208 °C associated with the change in crystal texture followed by an isotropic phase transition at 255 °C. But PDPPEMH crystals turn into an isotropic phase at 247 °C as compared to their di-hexyl counterparts. The average $p$-type mobilities for PDPP-MH and TDPP-MH was recorded to be 1.6 x 10^{-2} cm² V⁻¹ s⁻¹ and 0.3 cm² V⁻¹ s⁻¹, respectively. Whereas the average mobility values were two orders less for their di-hexyl analogues. The best mobility of 2 cm² V⁻¹ s⁻¹ was measured for TDPP-MH. The highest mobilities obtained from PDPP-DH and TDPP-DH were 3 x 10^{-3} cm² V⁻¹ s⁻¹ and 5 x 10^{-2} cm² V⁻¹ s⁻¹, respectively. On/off ratios of these devices were ~10^{-2}-10³. The current shows saturation behaviour with increase in source-drain voltage for almost all the devices prepared from DPP-MH but this was not observed for devices fabricated from DPP-DH. This may be related to the crystal quality or contact resistance at the semiconductor-electrode interface.

![Fig. 4 DSC thermograms of PDPP-MH and TDPP-MH](image1)

OFET devices were fabricated using Si/SiO₂ as a gate and dielectric material in bottom-gate bottom-contact (BGBC) device geometry. Photo-lithographically patterned gold (Au) source and drain contacts were developed on hexamethyldisilazane (HMDS) treated 190 nm thick dielectric layer of SiO₂. The solutions of respective organic materials were drop-casted on the substrate and crystals were grown by the slow evaporation method in the channel. The mono-hexyl derivatives form thin and sharp needle like crystal whereas their di-hexyl analogues form bigger crystallites in between the source and drain electrodes. The measurements were carried out at ambient condition at room temperature. The device data have been summarized in Table 1.

Significant differences in charge transport characteristics were observed between mono-and di-hexyl DPP derivatives, which are related to the intrinsic nature of the molecular semiconductors. We have observed higher $p$-type mobilities for mono-hexyl DPPs as compared to their di-hexyl counterparts. The average $p$-type mobilities for PDPP-MH and TDPP-MH was recorded to be 1.6 x 10^{-2} cm² V⁻¹ s⁻¹ and 0.3 cm² V⁻¹ s⁻¹, respectively. Whereas the average mobility values were two orders less for their di-hexyl analogues. The best mobility of 2 cm² V⁻¹ s⁻¹ was measured for TDPP-MH. The highest mobilities obtained from PDPP-DH and TDPP-DH were 3 x 10^{-3} cm² V⁻¹ s⁻¹ and 5 x 10^{-2} cm² V⁻¹ s⁻¹, respectively. On/off ratios of these devices were ~10^{-2}-10³. The current shows saturation behaviour with increase in source-drain voltage for almost all the devices prepared from DPP-MH but this was not observed for devices fabricated from DPP-DH. This may be related to the crystal quality or contact resistance at the semiconductor-electrode interface.

![Fig. 5 Output characteristics (a, b) and optical images of the corresponding devices (c, d) fabricated from single crystalline PDPP-MH (a, c) and TDPP-MH (b, d) microwires](image2)

Table 1 Summary of the OFET device parameters

<table>
<thead>
<tr>
<th>Compounds</th>
<th>Structure</th>
<th>$p$-type mobility ($\mu_{\text{average}}$) (cm² V⁻¹ s⁻¹)</th>
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<tr>
<td>PDPP-MH</td>
<td>Cofacial</td>
<td>1.6 x 10⁻²</td>
<td>10²</td>
</tr>
<tr>
<td></td>
<td>Layered</td>
<td></td>
<td></td>
</tr>
<tr>
<td>TDPP-MH</td>
<td>Cofacial</td>
<td>0.3</td>
<td>10²</td>
</tr>
<tr>
<td></td>
<td>Herringbone</td>
<td></td>
<td></td>
</tr>
<tr>
<td>PDPP-DH</td>
<td>Classical</td>
<td>2 x 10⁻⁴</td>
<td>10²</td>
</tr>
<tr>
<td></td>
<td>Herringbone</td>
<td></td>
<td></td>
</tr>
<tr>
<td>TDPP-DH</td>
<td>Classical</td>
<td>4 x 10⁻³</td>
<td>10³</td>
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Devices were fabricated in bottom-gate bottom-contact geometry.

The cofacial π-stacking improves the transfer integral as compared to edge-to-face herringbone packing according to Marcus theory, hence predicting better carrier transport through the π-stacked molecules if other factors remain the same.14, 15 Zhenan Bao and co-workers have also reported the same for tetracene derivatives.16 To correlate the experimental observation theoretically, the transfer integral ($\theta$) has been calculated for all the four crystals in the three crystallographic directions (Figure S15, S16, ESI†).1
Employing wB97xD function and 6-31g* basis set, the single point energy of dimeric structure (obtained from crystal structure) was calculated. The half of the energy difference between HOMO and HOMO-1 gives an estimate of transfer integral for hole transport while for electron transport, it was the energy difference between LUMO and LUMO+1. PDPP-DH has the t in z-direction, which corresponds to a herringbone arrangement. The t values (0.095 eV and 0.10 eV) predict equally good hole and electron mobilities. Along the y-direction smaller t (0.02 eV and 0.015 eV) indicates reduced transport. Charge transfer integrals are zero in x-direction. Removing one hexyl group (PDPP-MH), results in the cofacial crystal structure with hydrogen bonding in y-direction. However, very little (0.005 eV and 0.005 eV) t value was observed in this direction. Charge transfer integrals are about equally large in the other two directions, (0.05 eV and 0.07 eV (x-direction)) and 0.065 and 0.065 eV (z-direction) for hole and electron transport, respectively. But, for both mono-and di-hexyl TDPP, we observe that all the directions can take part in charge transport with relatively higher t value for DPP-MH system. The TDPP-DH system has the largest t (0.075 eV and 0.14 eV) along the y-direction predicting good electron mobility. Upon removing one hexyl group hydrogen bonding leads to rearrangement of the crystal structure. For TDPP-MH the best mobility is predicted in the z-direction, where t for electron and hole transport reaches 0.125 eV. The t values of all the four molecules in three crystallographic directions are summarized in Table S2 (ESI†).

Since, all the FET measurements were carried out in ambient condition, experimentally no electron transport was observed.

The overall result indicates that DPP-DH systems conduct mainly in one direction with high t value, while in DPP-MH systems charge transport occurs in two or three directions with the considerable t values. Therefore, the total charge transport predicted for mono-substituted systems is increased as compared to di-substituted ones. The hydrogen bonding direction does, however, contribute very little to the charge transport. Transport mainly occurs along π-stacks or along herringbone arrangements and appears to be influenced by the exact distance and orientation of the molecules.

Conclusions

In summary, two novel mono-hexyl DPP derivatives with phenyl and thiophene as donor groups have been synthesized and their properties are comprehensively characterized including single crystal X-ray analysis and charge transport measurement. The single crystal structure determination reveals the uniqueness of mono-hexyl DPP from their di-hexyl analogues. While di-hexyl PDPP and TDPP exhibit herringbone packing arrangements, the mono-hexyl derivatives show cofacial layered structure. The deviation is due to the intermolecular H-bonding by the free amide group in mono-hexyl DPP molecules. Higher p-type charge carrier mobilities were observed for PDPP-MH and TDPP-MH as compared to the di-hexyl DPP molecules. Theory agrees well with the experimental observation with respect to charge transport properties in this class of molecular semiconductors.

Notes and references