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## The route for commercial photoelectrochemical water splitting: a review of large-area devices and key upscaling challenges

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Green-hydrogen is considered a “key player” in the energy market for the upcoming decades. Among currently available hydrogen (H<sub>2</sub>) production processes, photoelectrochemical (PEC) water splitting has one of the lowest environmental impacts. However, it still presents prohibitively high production costs compared to more mature technologies, such as steam methane reforming. Therefore, the competitiveness of PEC water splitting must rely on its environmental and functional advantages, which are strongly linked to the reactor design, to the intrinsic properties of its components, and to their successful upscaling. This review gives special attention to the engineering aspects and categorizes PEC devices into four main types, according to the configuration of electrodes and strategies for gas separation: wired back-to-back, wireless back-to-back, wired side-by-side, and wired separated electrode membrane-free. Independently of the device architecture, the use of concentrated sunlight was found to be mandatory for achieving competitive green-H<sub>2</sub> production. Additionally, feasible strategies for upscaling the key components of PEC devices, especially photoelectrodes, are urgently needed. In a pragmatic context, the way to move forward is to accept that PEC devices will operate close to their thermodynamic limits at large-scale, which requires a solid convergence between academics and industry. Research efforts must be redirected to: (i) build and demonstrate modular devices with a low-cost and highly recyclable embodiment; (ii) optimize thermal and power management; (iii) reduce ohmic losses; (iv) enhance the chemical stability towards a thousand hours; (v) couple solar concentrators with PEC devices; (vi) boost PEC-H<sub>2</sub> production through the use of organic compounds; and (vii) reach consensual standardized methods for evaluating PEC devices, at both environmental and techno-economic levels. If these targets are not met in the next few years, the feasibility of PEC-H<sub>2</sub> production and its acceptance by industry and by the general public will be seriously compromised.

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### 1. Introduction

Renewable energy sources have been widely used by humankind for several millennia. In fact, until 250 years ago, hydro, wind, solar, and biomass were the only known and available energy sources.<sup>1</sup> After the Industrial Revolution in the 18th century, human society witnessed unprecedented technological development, with fossil fuels in the spotlight of energy production.<sup>2</sup> Energy demand never stopped growing, as the world population started experiencing significant growth, along with the improvement of living conditions.<sup>3,4</sup> As a consequence, energy consumption resulted in a progressively higher negative

impact on the environment.<sup>5,6</sup> Emissions of greenhouse gases (GHGs) have continuously increased due to the consumption of fossil fuels, leading to an accelerated global warming phenomenon with severe consequences for the climate stability of planet Earth, resulting in extreme and unpredictable weather events.<sup>7,8</sup> There is a global consensus to implement drastic changes in the way energy is harvested, stored and used, leading to an extensive reduction of GHG emissions aiming at keeping global warming below 1.5 °C relative to the pre-industrial times.<sup>9</sup> As humankind enters into the 4th Industrial Revolution, such a goal represents a great challenge to the scientific community, industry, and governments.<sup>10,11</sup>

Solar-powered technologies for producing chemical and electrochemical fuels are a clean, cost-competitive alternative to traditional fossil-fuel-based systems. Electrochemical solar hydrogen (H<sub>2</sub>) is particularly interesting since it allows the efficient harvest and conversion of abundant solar energy into a storable, transportable, environmentally friendly fuel. Green-H<sub>2</sub> is a valuable

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industrial chemical feedstock, especially for producing ammonia and steel and for CO<sub>2</sub> reduction into liquid synthetic fuels.<sup>12,13</sup> It can also be used in medium- and high-temperature heating processes in the industry for which electricity is not effective, or converted downstream into electricity for stationary or mobile applications.<sup>14</sup> These applications complement currently available technologies that capture solar energy and generate electricity (e.g. photovoltaics) or heat (e.g. solar-thermal systems).<sup>15</sup> At present, the most relevant processes for generating solar-H<sub>2</sub> are: (i) photovoltaic-electrolysis (PV-EC); (ii) photoelectrochemical (PEC) water splitting; (iii) photocatalysis; (iv) solar thermochemical processes; and (v) biological processes. Decoupled PV-EC is the available technology with the highest technological readiness level (TRL)

due to the maturity of both photovoltaic (PV) and electrolyser (EC) technologies, which have been optimized for decades.<sup>15</sup> However, PEC systems integrate light absorption and electrocatalysis in a single device and provide the opportunity to use thermal management for improving energy conversion efficiency.<sup>16,17</sup> The conversion of solar energy into chemical energy can also be accomplished *via* photocatalytic water splitting using particulate semiconductors, which is considered a promising technology with tremendous potential for large-scale applications, not only due to the readiness level on the photocatalyst synthesis but also due to the simple reactor designs.<sup>18</sup> Although the scope of the present review is not related to photocatalytic water splitting, one of the largest demonstrations of direct solar conversion into hydrogen is attributed to



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the research group of Domen, who successfully demonstrated a panel of 1 m<sup>2</sup>-size photocatalytic water-splitting working under natural sunlight with a solar-to-H<sub>2</sub> (STH) conversion efficiency of 0.4%.<sup>19</sup> The authors used a particulate RhCrOx/SrTiO<sub>3</sub>:Al photocatalyst in a one-step photoexcitation route.<sup>20</sup>

The successful implementation of PEC water splitting systems in the renewable energy market relies on their efficiency, durability, scalability, and cost-effectiveness. This route requires the optimization of device architectures for developing PEC systems that comply with the circular economy principles, comprising technical solutions already developed for high-performing laboratory cells.<sup>16,21</sup> The history of modern photoelectrochemistry is based on discoveries made in the mid-20th century on semiconductor surfaces.<sup>22–24</sup> Building upon these findings, notable advancements in photoelectrochemistry occurred in subsequent years, as is the case of the photoelectrocatalytic splitting of water. The original concept of PEC water splitting was first demonstrated by Boddy, in 1968,<sup>25</sup> who used an n-type titanium dioxide (TiO<sub>2</sub>) semiconductor to drive the water splitting reaction upon UV-light illumination and applying an external bias potential. However, it was the pivotal work of Fujishima and Honda, in 1972,<sup>26</sup> that boosted the research interest in the PEC water splitting field and different PEC devices and configurations have been studied since then. Thereafter, and over the last 50 years, PEC water splitting has been widely and exhaustively investigated. Unfortunately, the development and optimization of PEC devices addressing key upscaling challenges has received considerably less attention than materials development at lab-scale. Luckily, an ongoing change in paradigm is perceptible in the literature, with increasing relevance being given to the design and construction of large-area, or at least scalable, devices – Fig. 1. It is clear that the number of publications demonstrating large-area PEC devices almost doubled every five years for the past twenty-five years.

The STH conversion efficiency is a critical parameter to assess the competitiveness of PEC systems.<sup>28</sup> The United States Department of Energy has set an efficiency target of 25% to

reach market competitiveness of PEC water splitting systems.<sup>29</sup> The highest reported STH efficiency of 19.3% was achieved using multijunction monolithic PEC cells incorporating III–V semiconductors, but the technology is still far from commercialization due to its high cost and complexity.<sup>30</sup> Up to date, no single, earth-abundant and low-cost semiconductor material was identified to promote unbiased solar water splitting, delivering a high efficiency while operating stably on a time-scale of years. Due to the high potential difference needed to drive the overall water splitting reaction (typically >1.6 V), a dual-absorber tandem configuration, in which the bottom and top materials absorb complementary regions of the solar spectrum, has been increasingly investigated to enhance the generated photopotential.<sup>28</sup> To achieve STH efficiencies higher than 25%, pairing top absorbers with a bandgap range of 1.65–1.80 eV and bottom absorbers with a bandgap range of 0.95–1.15 eV is necessary.<sup>31</sup> The PEC–PV device, which combines the photoelectrode paired in tandem with a PV cell, envisaged as the most feasible tandem approach, surpassing the efficiency of PV–EC systems when using the same PV technology in both cases. As demonstrated by Rothschild and Dotan, a tandem PEC–PV system can be more competitive than a PV–EC system (using the same PVs in both cases).<sup>32</sup> By using a figure-of-merit (FOM) that considers the ratio between the STH efficiency of the complete system and the power conversion efficiency (PCE) of the PV module that drives the system, and then representing this FOM as a function of the fraction of the total power allocated to chemical power generation (assuming PV cells with a PCE of 20%, an electrolysis efficiency of 68.3% for the electrolyzers and PEC cells operating at the same bias potentials and current densities), the authors identified several points to boost the efficiency of PEC–PV tandem systems, *e.g.* through the use of power management and light management strategies.<sup>32</sup> Among the studied PEC–PV tandem configurations, special attention has been given to coupling a typical semiconductor in series with low-cost PV cells, including silicon (Si) PV solar cells, perovskite solar cells (PSCs), and dye-sensitized solar cells (DSSCs). As an example, tandem systems using a bismuth vanadate (BiVO<sub>4</sub>) photoanode were reported, displaying photocurrent densities close to the theoretical maximum limit.<sup>33–35</sup> Recently, the remarkable 17.6% STH efficiency was reported for a low-cost PEC–PV tandem cell comprising perovskite–silicon photocathode absorber.<sup>36</sup> These results show that there are ample opportunities for further improving the efficiency and reducing the costs of PEC systems, namely through the systematic optimization of each component individually. Along with performance indicators, the cost of produced H<sub>2</sub> is a critical metric for assessing the viability and future commercialization of the PEC–PV water splitting technology – Fig. 1. Pinaud *et al.*<sup>37</sup> estimated a levelized cost of H<sub>2</sub> (LCOH) from PEC systems between 4.10 USD and 10.40 USD per kg of H<sub>2</sub>, considering an STH efficiency of 10%, a lifetime of 10 years, and panel costs between 153 USD m<sup>−2</sup> and 316 USD m<sup>−2</sup>. Given an equivalent STH efficiency, Shaner *et al.*<sup>38</sup> concluded that utilizing an EC stack and a PV module would result in a higher cost of dispensed H<sub>2</sub>, mainly due to the fact that transporting H<sub>2</sub> from

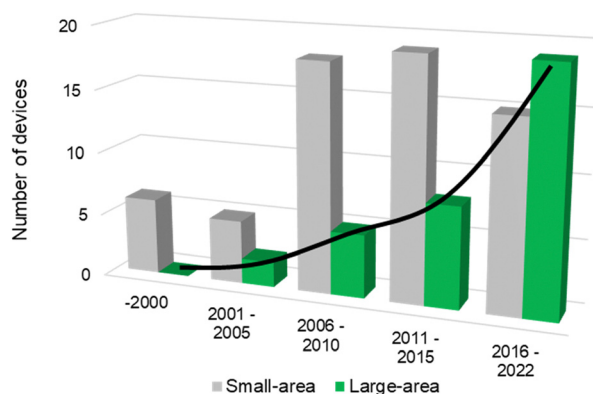


Fig. 1 Histogram of reported PEC reactors over the last 25 years concerning the number of small-area devices (grey) and large-area devices with an illuminated area higher than 50 cm<sup>2</sup> (green) – (searched keywords: “photoelectrochemical” + “water splitting” + “small area” + “large area” + “device”; database: Google Scholar, Scopus and SolarfuelsDB<sup>27</sup>).





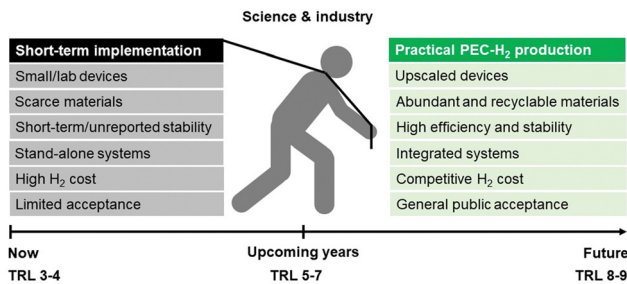


Fig. 2 Holistic pathway for commercial PEC-H<sub>2</sub> production and key challenges that must be addressed simultaneously by the scientific community and the industry.

the PV panels to compression units would be less expensive than transporting and conditioning electricity from the panels to the electrolyzers; still, the authors highlight that PEC systems still need great advancements, namely regarding the panel design and mounting to become truly competitive against grid-supplied PV-EC units.

As schematized in Fig. 2, a set of key requirements should be considered in fully integrated PEC-PV tandem systems. Apart from the performance of the photoabsorber materials and electrocatalysts, when commercialization is brought into sight, the PEC reactor itself, integrating all components, plays a crucial role in balancing the cost-effectiveness of such systems. It was established that an optimized PEC reactor must address the following requirements:<sup>39,40</sup> (i) maximum absorption of solar energy by the photoelectrodes and/or PV components (also under concentrated sunlight); (ii) electrolyte recirculation with an optimized internal flow pattern, assuring efficient heat dissipation and gas bubble removal; (iii) optimum arrangement of electrodes to minimize overpotential losses (ionic and electronic); (iv) efficient separation of the evolved gases; (v) comprise photo and chemically stable components; (vi) construction using highly recyclable components; (vii) use of inexpensive materials and low-cost maintenance; and (viii) scalability and easy integration in an H<sub>2</sub> production plant. A well-engineered PEC reactor that is suitable for large-scale PEC-PV systems must combine all these requirements.<sup>41</sup>

The following sections will address the progress on materials engineering for the main components of PEC systems, key design features of the main PEC device architectures, the challenges resulting from the use of concentrated sunlight and biomass derivatives, as well as the current status of PEC-H<sub>2</sub> production based on environmental and economic metrics.

## 2. PEC water splitting components

The field of materials remains the most challenging for addressing the key challenges for commercial PEC water splitting, mainly because: (i) for long-term PEC operation, the photoelectrode (PE), the counter-electrode (CE) and the membrane need to be chemically stable in the presence of aqueous electrolytes and evolved gases, as well as continuous electrolyte recirculation, exposure to normal and concentrated sunlight; and (ii) each component of the device must incorporate Earth-abundant,

non-toxic and low-cost materials, prepared by simple and scalable processes.<sup>42,43</sup> This section reviews promising materials used in the main components of a PEC device, focusing on low-cost and scalable fabrication techniques and strategies for improving their performance.

### 2.1. Photoelectrodes

PEC water splitting devices rely on photoactive electrodes, *i.e.* photoelectrodes, which have the ability to convert incident photons into electron-hole pairs ( $e_{CB}^- + h_{VB}^+$ ) in the conduction and valence bands, respectively, when exposed to sunlight.<sup>44</sup> This is only possible if photons' energy ( $E$ ),  $h\nu$  ( $h$  is the Planck's constant and  $\nu$  is the frequency), is equal to or higher than the semiconductor's energy bandgap ( $E_G$ );  $E_G$  is given by the difference between the bottom of the conduction band ( $E_C$ ) and the top of the valence band ( $E_V$ ).<sup>45</sup> When the semiconductor is in contact with an electrolyte whose redox energy ( $E_{Redox}$ ) differs from the Fermi level energy ( $E_F$ ), the equilibrium of the electrochemical potential is achieved by electron exchange at the interface between the semiconductor and the electrolyte. Electrons are transferred to the electrolyte, leaving in the semiconductor a depleted layer (space charge layer);<sup>44</sup> the Fermi level is then shifted due to the formation of an electric double layer at the electrode-electrolyte interface.<sup>46</sup> The depletion in electron concentration at the surface of the PE induces an electric field between the semiconductor's bulk and its surface, leading to the band bending phenomenon.<sup>44,47</sup> For a given semiconductor/electrolyte combination, there is a unique potential at which the potential drop between the surface and the bulk is zero, and there is no space charge layer – the flatband potential. For a certain redox pair, the flatband potential corresponds to the redox potential of the electrolyte when there is no excess charge in the semiconductor and, consequently, no band bending.<sup>48</sup> When a semiconductor, in equilibrium with an electrolyte, absorbs light, it has the capacity to change the band bending, since the electron and hole densities increase due to light absorption.<sup>44,47</sup> One of the key aspects concerning the choice of a semiconductor is related to its band edge positions, which determine the upper limits of reduction and oxidation potentials for the photogenerated electrons and holes.<sup>44</sup> The energy difference between the oxidizing hole and the reducing electron should be higher than the electrochemical potential required for water dissociation ( $> 1.23$  V).<sup>47,49</sup> An  $E_G$  of 1.6–2.0 eV, which covers the enthalpy of the reaction (1.48 V) and overpotentials, is considered to be the best compromise for a PE used in a PEC device.

Given these assumptions, the standardized metric for evaluating the overall performance of a PEC system is STH efficiency ( $\eta_{STH}$ ), defined as the amount of chemical energy produced, *i.e.* the rate of hydrogen production (mmol of H<sub>2</sub> s<sup>-1</sup>) against the solar energy input. The chemical energy produced is multiplied by the change in Gibbs free energy per mol of H<sub>2</sub> (at 25 °C,  $\Delta G = 237$  kJ mol<sup>-1</sup>). The solar energy input is the incident illumination power density ( $P_{in} = 0.1$  W cm<sup>-2</sup>, AM 1.5 G, considering standard PEC operating conditions) multiplied by



the illuminated PE area ( $A$ ,  $\text{cm}^2$ ). The  $\text{H}_2$  production rate can be directly measured by gas chromatography or mass spectrometry.

$$\eta_{\text{STH}} (\%) = \frac{\text{H}_2 \text{ Production rate} \times \Delta G}{P_{\text{in}} \times A} \quad (1)$$

Additionally,  $\eta_{\text{STH}}$  can also be determined using the relation that power is the product of potential, current, and the faradaic efficiency for  $\text{H}_2$  evolution ( $\eta_{\text{F}}$ ) as follows:

$$\eta_{\text{STH}} (\%) = \frac{P_{\text{op}}}{P_{\text{in}}} \times \eta_{\text{F}} = \frac{1.23 \times J_{\text{op}}}{P_{\text{in}}} \times \eta_{\text{F}} \quad (2)$$

where  $P_{\text{op}}$  is the operational power density, given by the product of generated photocurrent density ( $J_{\text{op}}$ , in  $\text{mA cm}^{-2}$ ) and potential at 1.23 V. From eqn (2), the operational photocurrent relates proportionally to the STH efficiency and is, therefore, a crucial parameter for improving the overall efficiency of a PEC device. However, maximizing the photocurrent of a PEC system implies the complementary improvement of the photopotential and fill factor of the PEs. In practice, PEC systems require an extra potential, *i.e.* a few hundred mV, to promote the water splitting reaction, mainly attributed to overpotentials linked with the water oxidation, charge transfer in the electrolyte and transport across the back contact of the semiconductor, electrode separation distance, and device geometry.<sup>50</sup> In PEC-PV tandem systems, the PV cell is electrically connected in series with the PE, *i.e.* it provides the extra bias needed for the water splitting. A conventional PEC device, in a three-electrode configuration setup, is represented in Fig. 3(a); the internal circuit is presented with more detail, enabling a more comprehensive analysis of electrical connections and flow of charged species – Fig. 3(b). For describing the basic working principles of a PEC cell, the energy band diagrams of a PEC cell based on a generic n- and p-type semiconductor, as well as a dual tandem configuration combining n- and p-type PE materials, are also shown in Fig. 3(c)–(e), respectively.

PEC devices designed for practical applications usually consist of a narrow reactor – the cell – where the electrolyte and both electrodes are inserted, typically assembled in separate compartments divided by an ion exchange membrane, which allows for charge balance while avoiding gas mixing. Front and/or back transparent windows allow sunlight to reach the surface of the photoactive materials.<sup>51</sup> Still, when someone addresses the topic of materials within the field of PEC- $\text{H}_2$  production, the component that instantly comes to mind is the photoabsorber material, *i.e.* the semiconductor or photoelectrode, as it causes the photon-to- $\text{H}_2$  conversion.<sup>52</sup> In fact, PEs are the key component that makes PEC cells more attractive compared to PV-EC systems, as they enable light absorption, charge generation and separation, hole, electron, ionic and molecular transport in a single device. Additionally, PEC devices can be operated at lower current densities, exhibiting lower overpotentials than commercial electrolyzers.<sup>53,54</sup>

An ideal semiconductor for PEC water splitting should meet the following key criteria: (i) good visible light absorption and narrow bandgap energy between 1.8 eV and 2.4 eV; 1.8 eV corresponds to the theoretical minimum bandgap for water

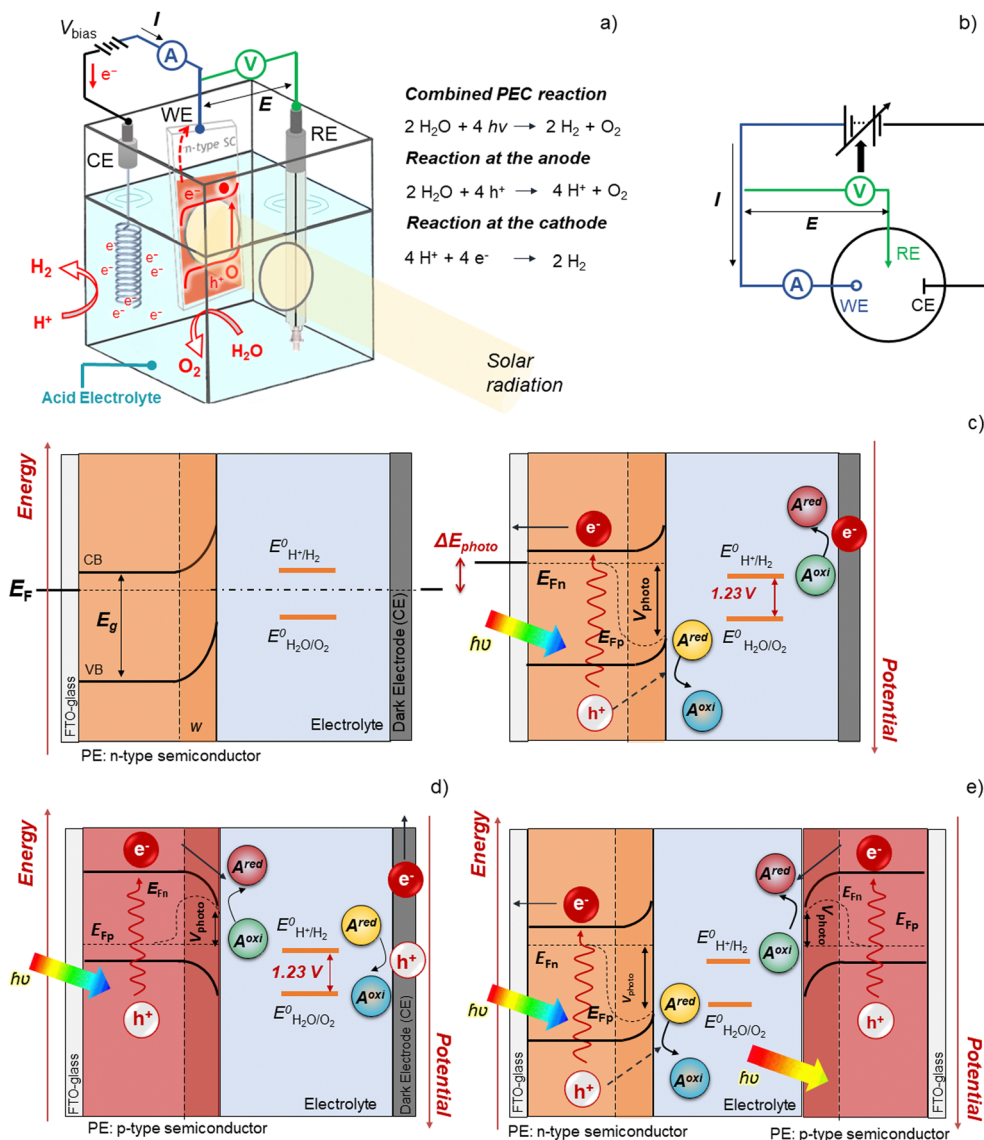
splitting of 1.23 eV plus the thermodynamic losses (0.3–0.4 eV) and overpotentials (0.4–0.6 eV) and 2.4 eV is the maximum bandgap value to achieve the target of 10% STH efficiency; (ii) suitable valence and conduction band edge positions, as the valence band should be below the redox potential for  $\text{O}_2$  evolution and the conduction band should be above the redox potential for  $\text{H}_2$  evolution; (iii) effective separation and transport of charge carriers, relying on both intrinsic characteristics (hole and electron mobility) and extrinsic properties (surface area, crystallinity, grain boundaries); (iv) high catalytic activity, essential for avoiding surface recombination and for decreasing the activation overpotentials; (v) chemical stability, which depends on the relative positions of the band edges and the respective decomposition potentials; and (vi) non-toxicity, Earth-abundance and low-cost fabrication.<sup>39,55</sup> So far, there is no single semiconductor reported that meets all these requirements.

The materials used to prepare photoelectrodes for PEC water splitting can essentially be divided into the so-called “classic” and emerging semiconductor materials.<sup>56–58</sup> Considering the short time window for making PEC water splitting competitive, conventional metal-oxide semiconductor materials should be preferable for optimizing large-area PEs, since they are easy to prepare and have already been deeply studied and demonstrated. Typically, transition metal-oxide PEs include  $\text{TiO}_2$ , hematite ( $\alpha\text{-Fe}_2\text{O}_3$ ), tungsten trioxide ( $\text{WO}_3$ ), bismuth vanadate ( $\text{BiVO}_4$ ), and cuprous oxide ( $\text{Cu}_2\text{O}$ ); tantalum nitride ( $\text{Ta}_3\text{N}_5$ ) and silicon-based PEs are also being highly reported. Among the emerging materials, organic, perovskite and kesterite semiconductors are three of the most studied.

**2.1.1. “Classic” photoelectrodes.** From the family of n-type semiconductors,  $\text{TiO}_2$  was the first material used to demonstrate PEC water splitting under simulated UV light.<sup>25,26</sup> This study encouraged the research work on PEC- $\text{H}_2$  production and several studies were reported in the following years using  $\text{TiO}_2$ . Still, its wide bandgap (3.2 eV for anatase and 3.0 eV for rutile) strongly restricts the maximum theoretical STH efficiency (1.3% for anatase and 2.2% for rutile), since only 5% of the solar spectrum (predominantly UV light) can be absorbed. This drawback has shortened the interest in developing efficient  $\text{TiO}_2$ -based PEs for solar water splitting.<sup>59</sup> Despite the valence band can be modified by incorporating non-metal species (such as carbon or nitrogen), and the conduction band can be changed *via* doping with 3d transition metal ions, no substantial change was observed in the bandgap and, consequently, no substantial enhancement was achieved in the STH efficiency.<sup>60,61</sup> As a result,  $\text{TiO}_2$  is no longer used as a classic material for PEs, being used instead as a doping agent or protective coating of other oxides.<sup>62,63</sup>

Hematite ( $\alpha\text{-Fe}_2\text{O}_3$ ) is one of the most promising photoanodes for solar water splitting and has been widely studied since the late 1990s. This metal-oxide presents remarkable chemical stability, low toxicity, and low-cost.<sup>64</sup> Also, it has a bandgap between 1.9 eV and 2.3 eV, which is nearly ideal for visible light harvesting, resulting in a maximum theoretical STH efficiency of *ca.* 16% (corresponding to  $12.6 \text{ mA cm}^{-2}$  at  $1.23 V_{\text{RHE}}$ ).<sup>59</sup> However, its use as a semiconductor for PEC- $\text{H}_2$





**Fig. 3** Conventional PEC device: (a) schematic representation of a generic PEC device, assembled in a three-electrode configuration, including the relevant reactions of photo-oxidation of water using an n-type photoelectrode. (b) Simplified schematic of a potentiostat attached to a PEC cell using working electrode (WE), working sense (WS), counter-electrode (CE), and reference electrode (RE) leads. The working electrode is typically the photoelectrode used for studying any electrochemical processes. The potential within the cell is measured between RE and WE. The current flow through both potential-sensing electrodes is minimized (ideally zero). The CE completes the electrical circuit. Current ( $I$ ) flows between CE and WE, while potential ( $E$ ) is read between WE and WS. Energy diagram of a PEC cell based on: (c) a photoanode (n-type semiconductor) and a dark electrode (metal cathode) in the dark (left side) and under illumination (right side); (d) a photocathode (p-type semiconductor) and a dark electrode (metal cathode) under illumination; and (e) a tandem arrangement, which combines a photoanode (n-type semiconductor) and a photocathode (p-type semiconductor) under illumination.

generation entails some drawbacks, including: (i) slow mobility of holes (*ca.*  $0.2 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ ), resulting in a very short diffusion length of 2–4 nm; (ii) poor surface water oxidation kinetics, leading to high surface recombination due to the accumulation of holes; and (iii) low absorption coefficient (on the order of  $10^3 \text{ cm}^{-1}$ ), requiring film thicknesses of at least 400–500 nm for full sunlight absorption.<sup>65,66</sup> Among the various reported techniques to prepare  $\alpha\text{-Fe}_2\text{O}_3$  PEs, solution-based methods have raised particular interest. The resulting PEs show different nanostructured morphologies, like nanorods, nanowires or worm-like protuberances, all with a high surface area. The

state-of-the-art  $\alpha\text{-Fe}_2\text{O}_3$  PE is shaped in nanowire arrays, prepared by a hydrothermal method, and is capable of delivering a record photocurrent density of  $6 \text{ mA cm}^{-2}$  at  $1.23 \text{ V}_{\text{RHE}}$  (*ca.* 7% STH efficiency);<sup>67</sup> this result corresponds to *ca.* 50% of the theoretical limit for this semiconductor. Still, nanostructured  $\alpha\text{-Fe}_2\text{O}_3$  presents stability challenges when compared with more compact structures, typically prepared by atomic layer deposition (ALD) or spray pyrolysis (SP). The most stable  $\alpha\text{-Fe}_2\text{O}_3$  PE reported to date is a 19 nm film prepared by SP, that is capable of generating  $1.05 \text{ mA cm}^{-2}$  at  $1.45 \text{ V}_{\text{RHE}}$  over 1000 h, under continuous simulated sunlight conditions, in





1 M KOH.<sup>64</sup> This preparation technique was successfully upscaled to a 50 cm<sup>2</sup> PE that generated *ca.* 0.45 mA cm<sup>-2</sup> at a bias potential of 1.6 V, over 1008 h under 1000 W m<sup>-2</sup>, and with electrolyte recirculation at 45 °C.<sup>43</sup> Indeed, this review's authors have consistently demonstrated the successful scale-up of flat and nanostructured  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> PEs over the years. Their achievements include scaling up to 25 cm<sup>2</sup> nanostructured hematite of porous nanocubes,<sup>68</sup> but also compact photoelectrodes up to 50 cm<sup>2</sup> and 100 cm<sup>2</sup>.<sup>43,69</sup>

Tungsten trioxide (WO<sub>3</sub>) is another n-type semiconductor that has been studied for PEC applications since the mid-1970s, presenting a good electron mobility (6 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup>) and a long charge carrier diffusion length (*ca.* 150 nm). Although this material has a bandgap of 2.7–2.8 eV, larger than  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, it is still capable of absorbing a reasonable fraction of the solar spectrum, with limited maximum theoretical STH efficiency (4.8%).<sup>70</sup> In contrast to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, it is stable only in aqueous acidic solutions (pH < 4), such as H<sub>2</sub>SO<sub>4</sub>, H<sub>3</sub>PO<sub>4</sub>, Na<sub>2</sub>SO<sub>4</sub> or CH<sub>3</sub>SO<sub>3</sub>H. There are several reported techniques for preparing low-cost and scalable nanostructured WO<sub>3</sub> PEs.<sup>71,72</sup> Sputtering allows obtaining photoanodes with good crystallinity, low intergrain and lattice defects; however, this technique is quite expensive and not easily scalable. Nanoporous or worm-like structures can be obtained by electrochemical anodization of metallic tungsten foils; the as-prepared PEs are opaque, and the roughness of the foil affects the growth shape and direction.<sup>73</sup> Electrodeposition allows the deposition of WO<sub>3</sub> films on glass substrates, but these PEs usually present low current densities (*ca.* 0.35 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>).<sup>74</sup> Overall, sol-gel remains the most common chemical process used to prepare WO<sub>3</sub> PEs; the resulting semiconductor layer is thick and highly porous. Reinhard *et al.* reported bare WO<sub>3</sub> PEs, prepared *via* a sol-gel method, consisting of consecutive depositions of a colloidal precursor solution by doctor-blading, followed by thermal annealing, which displays *ca.* 3.5 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>, remaining the state-of-the-art achievement.<sup>75</sup>

Bismuth vanadate is the most successfully demonstrated ternary metal oxide photoanode, being also a promising candidate for large-area PEC-H<sub>2</sub> production. This material has a bandgap of 2.4 eV, resulting in a theoretical maximum photocurrent density of 7.4 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>, corresponding to an STH efficiency of 9.1%. Similar to  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>, the performance of BiVO<sub>4</sub> photoanodes is mainly limited by the fast charge carrier recombination, due to the short hole diffusion length, which is estimated to be 70–100 nm; it is also limited by the slow water oxidation kinetics.<sup>76,77</sup> To overcome these limitations, BiVO<sub>4</sub> is often doped with Mo and W, which enhances the charge carrier-separation efficiency.<sup>78</sup> Spray pyrolysis is a common technique used to fabricate BiVO<sub>4</sub> photoanodes;<sup>77</sup> Ahmet *et al.* reported BiVO<sub>4</sub> photoanodes prepared by spray pyrolysis that achieved a photocurrent density of *ca.* 3.3 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>, with an onset potential of 0.4 V<sub>RHE</sub>.<sup>79</sup> BiVO<sub>4</sub> photoanodes are often used in hetero-junctions, usually with WO<sub>3</sub>, allowing photocurrent densities higher than 6 mA cm<sup>-2</sup> to be achieved at 1.23 V<sub>RHE</sub>, where nanostructured WO<sub>3</sub> is used as scaffold for the BiVO<sub>4</sub> top layer.<sup>76</sup> There are several reported techniques for fabricating

these heterostructures. Porous WO<sub>3</sub> structures (the scaffold) can be fabricated using colloidal templates, such as polystyrene spheres; the BiVO<sub>4</sub> precursor solution is later impregnated on the template followed by a heat treatment, which allows the removal of the polystyrene and crystallizing the doped BiVO<sub>4</sub>.<sup>80,81</sup> Another interesting approach consists of combining glancing angle deposition (GLAD) to fabricate WO<sub>3</sub> nanorods, followed by electrochemical deposition of BiVO<sub>4</sub>, which already allowed a PE displaying a photocurrent density of 6.72 mA cm<sup>-2</sup> to be fabricated at 1.23 V<sub>RHE</sub> (>90% of its theoretical limit).<sup>82</sup> The most stable BiVO<sub>4</sub> PE was demonstrated by Kuang *et al.*<sup>83</sup> These authors reported a particulate Mo:BiVO<sub>4</sub>/Ni/Sn electrode that generated *ca.* 2.6 mA cm<sup>-2</sup> at 0.6 V<sub>RHE</sub> over 1100 h, without noticeable performance losses after the complete test; stability measurements comprised eleven 100-h runs, with electrolyte replacement between any two runs and *in situ* self-regeneration of the catalyst. The photocorrosion inhibition was attributed to the specific high-temperature treatment; *in situ* self-generation of catalyst was assured by the thin Ni layer, which acted simultaneously as an ion source and electrical back contact.

From the family of oxynitrides, tantalum nitride (Ta<sub>3</sub>N<sub>5</sub>) has received special attention, due to its 2.1 eV direct bandgap and suitably positioned energy band edges for solar water splitting applications. Indeed, Ta<sub>3</sub>N<sub>5</sub> is probably the most promising semiconductor among the ones reaching photocurrents close to the theoretical limit of STH efficiency under sunlight. Liu *et al.* reported a multi-step strategy for improving the performance of Ta<sub>3</sub>N<sub>5</sub> grown on Ta metal foils, which yielded a maximum photocurrent density of *ca.* 12.1 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>, corresponding to 94% of its theoretical performance limit; this is an unprecedented achievement within the PEC-H<sub>2</sub> field.<sup>84</sup> The application of Ni(OH)<sub>x</sub>/ferrihydrite (Fh) layers onto Ta<sub>3</sub>N<sub>5</sub> proved to be efficient in mediating the interfacial charge transfer with Co and Ir catalysts, and a TiO<sub>x</sub> blocking layer for reducing the electron-hole recombination. Additionally, the Fh overlayer was also employed for protecting Ta<sub>3</sub>N<sub>5</sub> against photocorrosion.<sup>85</sup> Despite this impressive progress, Ta<sub>3</sub>N<sub>5</sub> still faces two critical shortcomings: poor stability and low photopotential. Fortunately, different strategies have been successfully employed to tackle these issues. First, stability can be improved, *e.g.* using electrolyte solutions containing a ferrocyanide hole scavenger; still, the reported 30 h of stable performance remains far from real application needs.<sup>86</sup> On the other hand, the low photopotential can be overcome by using tandem arrangements, which have the potential to generate sufficient photopotential to achieve water splitting with high STH efficiency. The most efficient Ta<sub>3</sub>N<sub>5</sub> PEs were prepared on opaque Ta foils, which limits the assembling options to front illumination only. To be used as front-side PE (back illumination), semi-transparent Ta<sub>3</sub>N<sub>5</sub> has to be fabricated on transparent conductive oxide (TCO) glass substrates, but they seem to reach lower photocurrents.<sup>87</sup> Higashi *et al.* fabricated a Ta<sub>3</sub>N<sub>5</sub> planar thin film grown on a transparent GaN/Al<sub>2</sub>O<sub>3</sub> substrate and decorated with an ultrathin NiFeOx electrocatalyst layer, generating 7.4 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>.<sup>88</sup> When paired in tandem with CuInSe<sub>2</sub> PV, an STH efficiency of 9% was obtained, which is the highest efficiency ever reported among metal-oxide/



nitride-based PEC–PV tandem cells; unfortunately, this device showed only 2 h of stable operation.

Spinel ferrites ( $\text{MFe}_2\text{O}_4$ ,  $\text{M} = \text{Zn, Mg, Cu, Ca, etc.}$ ) have also been the subject of several research studies on PEC water splitting. These ternary metal oxides are made of abundant materials, present a suitable bandgap for light harvesting in the visible range, *i.e.* 1.4–2.1 eV, and have displayed long-term photostability.  $\text{ZnFe}_2\text{O}_4$  (known as ZFO), with a bandgap range of 1.9–2.1 eV, enables broad absorption of visible light while presenting favourable chemical and morphological stability.<sup>89</sup> This material arose as a substitute for typically less efficient PEs, being frequently used as a secondary overlayer on composite PEs, such as  $\text{TiO}_2/\text{ZFO}$  or  $\alpha\text{-Fe}_2\text{O}_3/\text{ZFO}$ , boosting the charge extraction in the primary oxide.<sup>90</sup> However, as the main active photoanode material, the performance of ZFO has remained modest; among the best results, Xu *et al.* reported ZFO decorated AZO nanowires that generated a photocurrent of *ca.* 1.72  $\text{mA cm}^{-2}$  at 1.23  $V_{\text{RHE}}$  in a neutral electrolyte (0.1 M  $\text{Na}_2\text{SO}_4$ ).<sup>89</sup>

Cuprous oxide is the most studied photocathode for solar water splitting, being non-toxic, and abundant, presenting a bandgap of 2.0 eV, suitable for absorbing a significant part of the solar spectrum.<sup>91</sup> Despite having a maximum theoretical STH efficiency of 18%,  $\text{Cu}_2\text{O}$  has two main drawbacks that strongly limit its use for practical applications. First, this material is unstable under water-based solutions, as the reduction and oxidation potentials of monovalent copper oxide are located within the water splitting potentials, favouring photocorrosion.<sup>92</sup> Stability can be significantly improved by applying protective overlayers, such as  $\text{TiO}_2$  deposited by ALD; 50 h of stable performance was reported by Tilley *et al.*<sup>93</sup> Also, combining  $\text{Cu}_2\text{O}$  with an n-type semiconductor with a more positive conduction band, which forms a p–n junction that promotes a faster transfer of electrons from  $\text{Cu}_2\text{O}$  to the n-type semiconductor, allows improving both stability and efficiency. The second major drawback of  $\text{Cu}_2\text{O}$  is related to its charge carrier diffusion length.  $\text{Cu}_2\text{O}$  films typically are  $>1 \mu\text{m}$  thick to efficiently absorb sunlight; however, the electron diffusion length is limited to about 200 nm, which hinders the collection of photo-generated carriers. Accurate morphology control, based on advanced preparation techniques that allow obtaining nanostructured PEs, is the most attractive approach to address this challenge. A good example was implemented by Pan *et al.*,<sup>35</sup> reporting  $\text{Cu}_2\text{O}$  photocathodes able to deliver an unprecedented photocurrent density of 10  $\text{mA cm}^{-2}$  over 100 h of continuous operation, which was a state-of-the-art result, but still far from the ultimate target STH efficiency of 25%, set by the Department of Energy.<sup>29</sup>

Silicon-based PEs, the most common semiconductors in the PV industry, have also been widely used for PEC water splitting; the bandgap of 1.1 eV matches well with the solar spectrum, presenting a theoretical maximum photocurrent of 44  $\text{mA cm}^{-2}$  under 1-sun. However, bare Si PEs undergo fast oxidation when in contact with aqueous electrolytes. This challenge can be overcome by applying protective layers that passivate the surface states of Si. As an example, the alignment of  $\text{TiO}_2$ 's conduction band with both the hydrogen evolution reaction (HER) and with Si enables its use as a protective overlayer; additionally,  $\text{TiO}_2$  facilitates electron transfer from Si with negligible resistance.<sup>94</sup>

Inspired by the low-cost and high open-circuit potential difference displayed by amorphous Si (a-Si) solar cells, Lin *et al.* reported a Si photocathode coated with a  $\text{TiO}_2$  protection layer, and with Pt or Ni–Mo co-catalysts that generated 10  $\text{mA cm}^{-2}$  at 0.6  $V_{\text{RHE}}$  in 0.5 M KHP (potassium hydrogen phthalate), for 12 h of continuous operation.<sup>95</sup> In 2020, Fu *et al.* reported a bifacial Si photocathode coated with a semi-transparent 5 nm Pt layer that showed an unprecedented photocurrent density of 61.2  $\text{mA cm}^{-2}$  at 0  $V_{\text{RHE}}$  under bifacial 2-sun illumination; this system enables harvesting sunlight both at the front and at the back sides, yielding *ca.* 57% of excess  $\text{H}_2$  when compared with monofacial PEC systems, presenting a stable performance over 370 h in 1 M  $\text{H}_2\text{SO}_4$ .<sup>96</sup> Despite being largely used in first generation PV cells, it is important not to forget that metallic Si is listed as critical raw material (CRM), so its use must be greatly reduced in the next few years, including in PEC applications.

Fig. 4(a) shows the number of publications on some of the most studied semiconductors over the past decades, namely  $\alpha\text{-Fe}_2\text{O}_3$ ,  $\text{BiVO}_4$ ,  $\text{WO}_3$ ,  $\text{Cu}_2\text{O}$  and  $\text{Ta}_3\text{N}_5$ . Fig. 4(b) shows the state-of-the-art of these PE materials, based on the three-pillar criteria for commercialization: scalability, stability, and efficiency. Hematite  $\alpha\text{-Fe}_2\text{O}_3$  has been the most studied material, but publications on  $\text{BiVO}_4$  have increased exponentially in the last 5-year period, followed by  $\text{Cu}_2\text{O}$ ,  $\text{Ta}_3\text{N}_5$ , and  $\text{WO}_3$ . Nevertheless, it is clear that efficiency has received more attention among the published research studies. So far,  $\alpha\text{-Fe}_2\text{O}_3$  is the semiconductor material that shows the best compromise among the three key criteria, although there is still no  $\alpha\text{-Fe}_2\text{O}_3$  PE capable of fully address simultaneously these targets. Nonetheless, some published studies have shown the potential of this semiconductor material to achieve important halfway targets for more than one pillar criterion simultaneously, such as an active area of  $\geq 50 \text{ cm}^2$  and a stable performance for more than 1000 h.<sup>43</sup> Overall, these results highlight the complexity of preparing large-area PEs that are simultaneously efficient and stable; it is important to emphasize that the process of discovering functional PE materials typically takes a minimum of 10 years.

**2.1.2. Emerging photoelectrodes.** In more recent years, other interesting types of PEs have emerged following the advancements in the PV field. Organic semiconductors are a particular type of emerging material that has aroused interest for PEC applications. Organic semiconductors are essentially polymers or small molecules with extended pi-conjugation of carbon bonds, presenting superior charge-transfer characteristics to their inorganic counterparts, and the possibility of being used both as photoanodes or photocathodes. The potential of semiconducting organic materials to transport electric current and absorb light in the UV-Vis part of the solar spectrum comes mainly from the  $\text{sp}^2$  hybridization of carbon atoms. Over the past decades, organic semiconductors have been successfully used in PV applications, namely in DSSCs. Due to the particularities of organic semiconductors, organic-PV devices present unique advantages such as the possibility of being applied onto flexible substrates, semi-transparency, low-cost manufacturing, and aesthetics, enabling their use in building-integrated photovoltaic (BIPV) and agrivoltaic applications.<sup>104</sup> Also, most organic





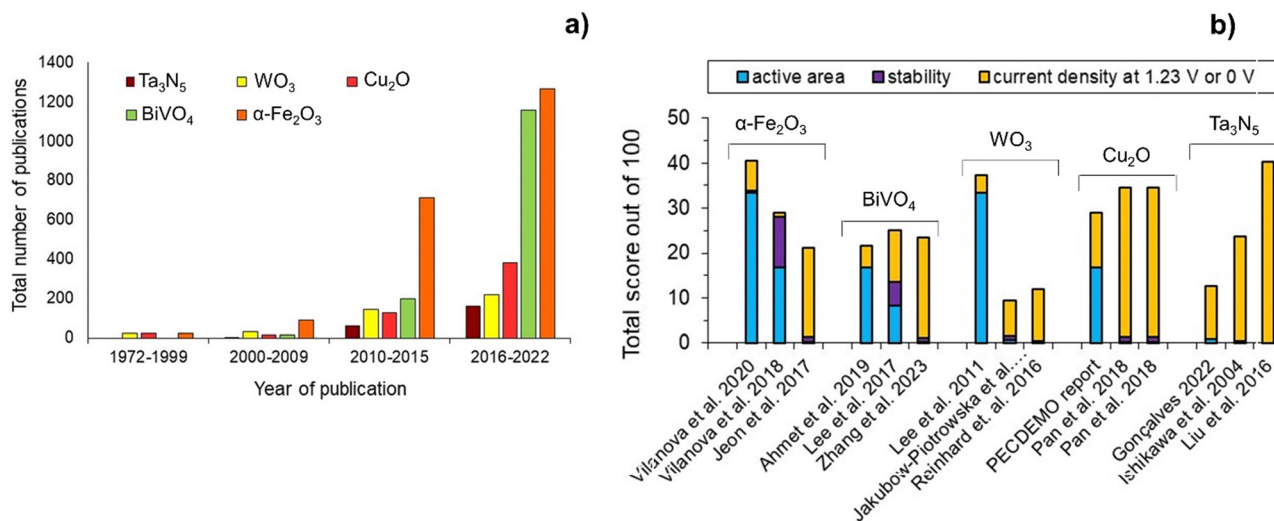


Fig. 4 Overview of the most studied semiconductors for PEC water splitting: (a) number of publications over the past decades (database: Google scholar and SolarfuelsDB<sup>27</sup>); and (b) performance ranking of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>,<sup>43,67,97</sup> BiVO<sub>4</sub>,<sup>79,98,99</sup> WO<sub>3</sub>,<sup>75,100,101</sup> Cu<sub>2</sub>O<sup>35,102</sup> and Ta<sub>3</sub>N<sub>5</sub><sup>84,86,103</sup> PEs based on active area, stability and current density generated at a reference potential of 1.23 V (except 0 V for Cu<sub>2</sub>O); normalization considered equal 33.3(3)% contribution for each criterion, delivering a total score out of 100, and considering the following targets: (i) active area of 100 cm<sup>2</sup>; (ii) 2920 h of stable performance (8 h over 365 days; max. 10% current loss); and (iii) current density of 10 mA cm<sup>-2</sup>.

semiconductors used in DSSCs are solution-processable, which enables large-area fabrication, namely *via* simple ink-based printing techniques.<sup>105</sup>

These advantages can be a differentiating aspect in PEC applications if successfully replicated. Nevertheless, the use of organic semiconductors in PEC devices has remained barely addressed, mainly due to their poor stability in aqueous solutions.<sup>106</sup> Early organic PE materials included polypyrrole, polyacetylene, polythiophene derivatives, poly(ethylenedioxythiophene) (PEDOT) derivatives, polyaniline, polythiophene-based photocathodes, poly(thienothiophene) (PTT) and poly(dithienothiophene) (PDTT).<sup>107</sup> Given the huge variety of organic semiconductors, they can essentially be grouped in dye-sensitized and thin films.<sup>105</sup> Since the first demonstrations of efficient proton reduction using organic PEs, research has been focused on the buried-junction approach, especially on introducing an anti-corrosion layer, and on developing highly tunable non-fullerene acceptors and wide-bandgap donor polymers.<sup>108</sup> Still far from the PEC performance of other semiconductor materials, the photocurrent density generated by organic photocathodes in PEC water-splitting devices has increased from a few  $\mu\text{A cm}^{-2}$  in acidic solutions at high negative potentials to over 100  $\mu\text{A cm}^{-2}$  in neutral solutions at low/no applied potentials.<sup>107</sup> Indeed, the rapid advances within the organic semiconductor field for solar water splitting have provided a vast number of high-performance donor polymers and non-fullerene acceptors with carefully tailored chemical structures, allowing them to reach high photocurrent and open circuit potential. As the band positions and bandgaps of organic semiconductors can be easily tuned, in multijunction devices, by combining two or more sub-cells with complementary bandgaps and connecting them in series, the total open circuit potential ( $E_{\text{OC}}$ ) can theoretically be sufficient to drive water

splitting, resulting in a maximum STH efficiency of up to 30%, avoiding the need of coupling an external PV cell, which is often considered an essential requirement for PEC water splitting feasibility.<sup>109</sup> Since water oxidation is the limiting half-reaction, and given that many organic materials are easily degraded upon oxygen exposure, organic photocathodes are considered more promising than organic photoanodes.<sup>110</sup> Even so, high-performance and stable organic materials were already reported. As an example, Yu *et al.* used nickel foils, GaIn eutectic, and layered double hydroxides to passivate the photoactive layer of p-type polymers and n-type non-fullerene materials.<sup>111</sup> These organic photoanodes generated 15.1 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub> and displayed an onset potential of 0.55 V<sub>RHE</sub>, resulting in an STH efficiency of 4.33%. These photoanodes were able to maintain 90% of the initial photocurrent for 10 h, when continuously operated at 1.3 V<sub>RHE</sub>. Zhang *et al.* have demonstrated a robust organic photocathode comprising a bulk heterojunction (BHJ) and an organic charge-selective layer.<sup>112</sup> This optimized photocathode generated *ca.* 4 mA cm<sup>-2</sup> at 0 V<sub>RHE</sub>, retaining 90% of the initial performance over 6 h. Combining the optimized BHJ photocathode with a benchmark BHJ photoanode, the authors then demonstrated a 2.4 cm<sup>2</sup> organic PEC tandem cell for complete solar water splitting, capable of generating a photocurrent of 0.6 mA cm<sup>-2</sup> in a two-electrode configuration, resulting in an STH efficiency of 0.3%. Finally, organic-based PEs include the so-called metal-organic framework-based materials (MOFs). These materials essentially comprise heterostructures used to fabricate thin films, where the MOF layer is used as a sacrificial template (Ti-based, Fe-based, Zn-based, and Co-based) or as a coating.<sup>113</sup> An example of a MOF-based photoelectrode displaying good PEC performance was reported by Pan *et al.*; the authors used NiFe-metal-organic frameworks (NiFe-MOFs) as a protective layer and cocatalyst on oxygen-vacancy-rich BiVO<sub>4</sub>



(Ov-BiVO<sub>4</sub>), forming a core-shell structure that produced 5.3 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>.<sup>114</sup>

Since the early 2010s, metal halide perovskites have emerged as one of the most promising materials for low-cost and efficient third-generation PV cells.<sup>115</sup> These materials comprise a polycrystalline structure with the chemical formula ABX<sub>3</sub>, with A referring to methylammonium (MA), formamidinium (FA), or cesium (Cs), B referring to Pb or Sn, and X referring to the halide atom, *i.e.* I<sup>-</sup>, Br<sup>-</sup>, or Cl<sup>-</sup>, or combinations of them.<sup>3</sup> As photo-absorber materials for PV applications, perovskites exhibit remarkable optoelectronic properties, namely high absorption coefficients, high electron and hole mobilities and diffusion lengths, superior structural defect tolerance and shallow point defects, low surface recombination rate and favorable grain boundaries, which have led to an unprecedented increase in efficiency, from 3.8% in 2009 to the present 26.1%.<sup>116</sup> Like in PV applications, perovskite materials, in this case perovskite oxides with the chemical structure ABO<sub>3</sub>, became highly attractive for PEC applications, offering better optoelectronic properties than simple oxides, mainly related to superior structural flexibility, high electrocatalytic activity, better sunlight absorption capability and simultaneous control of bandgaps and band edges.<sup>117</sup> For practical PEC water splitting, perovskite materials can be used as the main photoelectrode, *i.e.*, in contact with the electrolyte solution, or used in a tandem arrangement. In fact, Karuturi *et al.* achieved a remarkable STH efficiency of 17.6% for self-driven solar water splitting by pairing a semitransparent perovskite with a high bandgap (1.75 eV) with a p<sup>+</sup>nn<sup>+</sup>-Si/Ti/Pt photocathode capable of generating a current density of 39.7 mA cm<sup>-2</sup> at 0 V<sub>RHE</sub> and an onset potential of 590 mV.<sup>36</sup> The Reiser group has also been developing interesting solutions that use perovskites for solar water splitting, either as main PE or as dual absorbers in tandem arrangements.<sup>118,119</sup> One of the most relevant contributions was reported in 2022, where the authors successfully demonstrated the fabrication of scalable lightweight artificial leaves made of lead halide perovskite photocathodes deposited onto indium tin oxide-coated polyethylene terephthalate, comprising thin, flexible substrates and carbonaceous protection layers. This lightweight perovskite-BiVO<sub>4</sub> PEC device achieved an STH efficiency of 0.58% and its scalability was demonstrated using a 100 cm<sup>2</sup> stand-alone device, which sustained stable performance for 24 h.<sup>120</sup> Still, to be used as the main photoelectrode in conventional PEC device architectures, perovskite oxides have to be stable to the highly demanding operating conditions, namely to water exposure, temperature, and mechanical erosion promoted by gas bubbles evolution and electrolyte recirculation.<sup>121</sup> Possible ways to overcome such adversities include the use of thin overlayers and encapsulation. An example of the successful application of a thin protective overlayer was demonstrated by Liu *et al.*<sup>122</sup> The authors have reported porous perovskite photoanodes capable of being operated in standard PEC conditions, *i.e.* in contact with an aqueous electrolyte. The BiFeO<sub>3</sub> photoanodes, covered with an ultra-thin TiO<sub>2</sub> overlayer (5–6 nm) and a cobalt oxide/oxyhydroxide co-catalyst, generated 0.16 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub> in 1 M NaOH, achieving an onset potential of 0.6 V<sub>RHE</sub>. On the

other hand, Chen *et al.* have successfully demonstrated encapsulated perovskite-based PEs for solar water splitting. In this case, the device layout consisted of two monolithically encapsulated perovskite (FA<sub>0.80</sub>MA<sub>0.15</sub>CS<sub>0.05</sub>PbI<sub>2.55</sub>Br<sub>0.45</sub>) solar cells, covered with CoP and FeNi(OH)<sub>x</sub> co-catalysts, working as a photocathode and a photoanode, respectively. The CoP-based photocathode generated 12.4 mA cm<sup>-2</sup> at 0 V, whereas the FeNi(OH)<sub>x</sub>-based photoanode generated 11 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>, resulting in an unassisted overall STH efficiency of 8.54%. The encapsulation was made using a thin polyisobutylene (PIB) layer and a cover glass, enabling these electrodes to be in direct contact with the electrolyte (0.5 M NaOH) and presenting a stable performance over 13 h of continuous operation.<sup>123</sup> Another remarkable achievement using perovskite PEs for PEC water splitting was reported by Daboczi *et al.* The authors reported a CsPbBr<sub>3</sub>-based photoanode that displayed a 0.4 V<sub>RHE</sub> onset potential and generated 8 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>, being capable of retaining 100% of its stabilized photocurrent density after 100 h of continuous operation. Enhanced charge transfer was attributed to the phase engineering of CsPbBr<sub>3</sub> and to the NiFeOOH catalyst applied on a protective graphite sheet.<sup>124</sup>

Kesterite (Cu<sub>2</sub>ZnSn(S,Se)<sub>4</sub>, CZTS) materials are another class of emerging materials for PEC applications.<sup>63</sup> Kesterites have been primarily studied for PV applications, in part because they are abundant, low-cost, and non-toxic. The interest in CZTS materials for PEC-H<sub>2</sub> production has arisen due to the capability of changing the bandgap energy from 1.0 eV to 1.5 eV – or even higher – depending on the used elements, chemical reactivity, and the very high coefficient of light absorption, among others. As a p-type semiconductor with suitable optoelectronic properties (absorption coefficient of *ca.* 10<sup>4</sup> cm<sup>-1</sup> and energy bandgap of *ca.* 1.4–1.5 eV), CZTS can be used as a photocathode for H<sub>2</sub> evolution.<sup>125</sup> Yokoyama *et al.* first demonstrated the application of CZTS as a photocathode in 2010.<sup>126</sup> Surface modifications, *i.e.* CdS and TiO<sub>2</sub> layers, were implemented to accelerate charge transfer and to protect the lower layer from corrosion in the electrolyte solution. Ros *et al.* were able to demonstrate the feasibility of CZTS/Se PEs in highly acidic conditions (pH < 1), with no detected degradation, by applying a protective TiO<sub>2</sub> layer deposited by ALD, achieving a current density of 10 mA cm<sup>-2</sup> at 0 V<sub>RHE</sub>.<sup>127</sup> However, kesterite-based PEC systems still lack adequate demonstration in high-performing and upscaled devices, mainly due to limitations arising from (1) performance and stability problems; (2) misalignment of energy bands and interface defects; (3) the lack of compatible surface catalysts; and (4) the complexity of kesterite-based architectures for high-performing and stable devices. Similar to PV applications, the performance of CZTS highly depends on the buffer layer, as it forms a p–n junction that facilitates the separation of photo-induced electron–hole pairs by the built-in electric field of the junction. Therefore, in CZTS PEs, having a charge-selective contact layer is essential for the efficient extraction of photoexcited charge carriers. Among many major sources of interfacial recombination in these materials, the energy band misalignment and recombination through interfacial defects can be pointed as the major ones.



Also, elemental interdiffusion at the interface affects the carrier transport. Charge-separating contacts with favorable energy band-alignment using interface passivation strategies can be a possible solution, as it has already demonstrated promising results in PV applications. For that, the conduction band position of buffer layers (e.g. CdS, Zn(O,S) and TiO<sub>2</sub>) can be tuned for better band matching. Other buffer layer parameters should also be optimized for optimum transparency and conductivity, charge carrier lifetime, and extraction. Surface passivation strategies include chemical and thermal treatments of the kesterite/buffer layer. Doping has also been proven to be an effective strategy for improving the PEC performance of CZTS PEs. As an example, Si-doping has been demonstrated to improve the generated photocurrent by 2.39 times, achieving *ca.* 5.57 mA cm<sup>-2</sup> at -0.41 V<sub>RHE</sub> in a CZTSSiS/CdS/ZnO PE.<sup>128</sup>

**2.1.3. Key strategies to improve the performance of photoelectrodes.** Most studies on the preparation of efficient PEs used for PEC-H<sub>2</sub> production follow a common approach for improving their performance, based on three main pillars: (i) morphology control; (ii) doping; and (iii) surface control.<sup>66,129</sup>

Morphology control has been widely used for increasing the photocurrent density generated at lower potentials; it overcomes the trade-off between enhancing the optical absorption and shortening the diffusion length of minority carriers, while improving the specific surface area, essential to drive the redox reactions.<sup>130</sup> Indeed, many different preparation techniques have been described for obtaining semiconductors with specific morphologies.<sup>50,131,132</sup> On the one hand, nanoparticles, dendrites, and mesoporous structures suffer from recombination and poor transport across the semiconductor, especially due to grain boundaries. Nanostructures such as worm-like protrusions, nanowires, nanotubes and nanorods, which have high aspect ratio and large surface area, enhance charge carrier collection by minimizing recombination at grain boundaries; in nanostructures with small diameters, the distance needed for the holes to diffuse to the semiconductor–electrolyte interface is shortened, thus mitigating the bulk recombination.<sup>66</sup> Typically, thick and porous morphologies have a higher surface area and are capable of generating higher photocurrent densities; however, they usually present low stability and the resulting photoanodes are opaque (which is a drawback when multi-layer approaches or tandem arrangements are envisioned). Also, thicker semiconductor layers have increasing recombination problems.<sup>50</sup> On the other hand, compact planar films, usually relatively thin (<40 nm), present high transmittance and assure the full coverage of the underneath conductive substrate, being therefore remarkably stable.<sup>133</sup> However, due to the small specific interfacial area, these morphologies present lower performance, typically photocurrent densities <1.5 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>.<sup>64,134</sup>

The incorporation of doping agents in the semiconductor lattice introduces additional majority carriers and creates steeper potential gradients (thinner space-charge layers), therefore improving charge separation and reducing the electron–hole recombination rate.<sup>135,136</sup> Still, doping can decrease the particle size and crystallinity of the hosting semiconductor, increasing bulk recombination and forcing charge carriers to travel longer

distances to reach the current collector.<sup>137</sup> There are different reported strategies for incorporating dopants in the semiconductor lattice.<sup>138,139</sup> Intrinsic doping is the simplest approach, such as *via* high-temperature annealing of semiconductors prepared over FTO glass substrates (normally over 650 °C), which enables the migration of Sn from the FTO layer to the main lattice of the semiconductor.<sup>140,141</sup> Other strategies usually comprise the addition of the doping agent to the precursor. However, there is no standardized procedure for choosing the more suitable doping agent, or the optimal doping level, since the beneficial effect of each doping agent is highly dependent on both the intrinsic properties of the semiconductor and on its morphology. As an example, the addition of Ti as a dopant to α-Fe<sub>2</sub>O<sub>3</sub> precursors must be carried with utmost precision since slightly too high concentrations may lead to the complete vanishing of the photoactivity. In the specific case of α-Fe<sub>2</sub>O<sub>3</sub>, several doping agents can be used to enhance its performance, namely Si, Ti, Sn, Nb, Zr, Pt, Mn, Zn and Ni; from these dopants, Sn is more effective for thin planar films, while Ti plays a more significant role on porous nanostructures.<sup>142</sup> For highly doped semiconductors, one possibility is to introduce a gradient in the dopant concentration across the film; as an example, doping BiVO<sub>4</sub> with a gradient of tungsten atoms allows the creation of a distributed n<sup>+</sup> n<sup>-</sup> homojunction with a corresponding electric field that enhances charge separation.<sup>78</sup>

The surface of the semiconductor, in contact with the electrolyte, requires special attention when improving the photocurrent density generated at lower potentials. In a semiconductor–liquid junction, the rate of charge transfer at the interface relates to the rate of carrier generation and separation in the semiconductor, which makes surface states critical to form the depletion region that influences charge transport. Therefore, surface states located on truncated sites may act as charge-trapping spots, recombination centres and Fermi level pinning states, being highly detrimental to the overall performance. Chemical etching is a simple and effective technique for improving the oxygen evolution reaction (OER), promoting the interfacial charge transfer, and inhibiting surface recombination, by changing the chemical state of the semiconductor surface. A good example of this strategy was reported by Cao *et al.* for α-Fe<sub>2</sub>O<sub>3</sub> PEs;<sup>143</sup> a synergetic cathodic shift of 100 mV in the photocurrent onset potential and an increase of the photocurrent density from 1.2 mA cm<sup>-2</sup> to 1.6 mA cm<sup>-2</sup> measured at 1.23 V<sub>RHE</sub> were demonstrated and attributed to surface corrosion promoted by acid treatment over 90 min (with an aqueous solution of HCl). Although no obvious change was observed in the surface morphology, electrochemical impedance spectroscopy (EIS) analyses showed that the interface charge transfer rate increased after surface corrosion, indicating that back-reaction losses were mostly suppressed.

The deposition of co-catalysts is another surface treatment that allows reducing surface states, enhancing the performance of a PE, mainly at more cathodic potentials. Co-catalyst materials can play different roles at the semiconductor–liquid interface, as they can: (i) facilitate the bond-making and bond-breaking reactions; (ii) passivate recombination sites; (iii) tune the band





structure energetics; and (iv) act as a protective layer against corrosion.<sup>144</sup> Catalysts can be applied as an impermeable layer, turning the semiconductor–liquid junction into a semiconductor–catalyst junction. Therefore, the band bending is no longer governed by the electrolyte redox potential but becomes dependent on the work function difference at the semiconductor–catalyst interface. The band edge positions can then change depending on the chosen catalyst, which may allow overcoming unfavourable band edge positions. Depending on the type of application, co-catalysts are usually divided into three main categories: (i) noble metals; (ii) transition metals; and (iii) non-metals and metalloids. Among all, noble metals such as iridium and ruthenium for the OER and platinum for the HER are among the best-performing catalysts; however, these are critical raw materials.<sup>55,145,146</sup> For minimizing their use, the surface-to-bulk atomic ratio can be increased without compromising efficiency and with substantial gains in terms of metal utilization. Dias *et al.*<sup>140</sup> reported the use of IrO<sub>2</sub>/RuO<sub>2</sub> as an OER catalyst, deposited on  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> thin films, to reduce overpotentials and improve photopotential. These authors achieved a 140% increase in photocurrent density at 1.23 V<sub>RHE</sub>, compared with a bare  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> PE and a final photopotential of 1.20 V, the highest value reported for  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>-based PEs. Due to the scarcity, and high cost of Ir-based electrocatalysts, several new approaches have been conducted aiming to lower this noble metal content without impairing performance, such as dispersing Ir oxides on highly conductive, stable, and with large specific surface area supports.<sup>147</sup> This strategy was mainly reported for water electrolysis but it can be used for PEC water splitting as well. Non-noble metal oxides such as antimony-doped tin oxide (ATO),<sup>148</sup> indium tin oxide (ITO),<sup>149</sup> and fluorine-doped tin oxide (FTO)<sup>150</sup> are examples of these supports. Alternatively, transition metals and metalloids have been studied as co-catalysts in PEC systems. Despite presenting a slightly lower activity than noble metal-based catalysts, these elements are cheaper and more abundant, being a more sustainable approach for industrial applications. Over the past decade, co-catalysts such as cobalt phosphate (Co–Pi), Fe- or Ni-oxides have been widely studied for passivating surface states on different semiconductors.<sup>151</sup> As an example, Co-based catalysts can be used to assist photogenerated holes to move away from recombination sites, located at the surface,<sup>152</sup> however, Co-based materials bring special environmental concerns, which can significantly hinder their applicability. Nickel oxides (NiO<sub>x</sub>) have received particular attention due to their remarkable stability and functionality.<sup>153</sup> The use of NiO<sub>x</sub> overlayers on BiVO<sub>4</sub> photoanodes promotes an efficient charge transfer to the surface, thus reducing charge recombination and improving OER kinetics. In a study by Zhang *et al.*, NiO<sub>x</sub>-modified/Mo:BiVO<sub>4</sub> photoanodes generated a photocurrent density of 2.44 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>, which is a 2-fold increase compared to uncatalyzed samples.<sup>153</sup> Some co-catalysts can also act as a protective overlayer, when applied as a thin uniform coating, acting as a physical barrier against severe reaction conditions.<sup>154,155</sup> An example of this approach is the use of a TiO<sub>2</sub> overlayer on Cu<sub>2</sub>O and PEs.<sup>93</sup> Nevertheless, the protective effect can only be accomplished if the catalyst coating is highly

stable, and transparent, to avoid compromising light-harvesting by the semiconductor layer.<sup>94</sup> When catalyst films are not applied uniformly, the STH efficiency can be hindered, since most catalysts have a high extinction coefficient (opaque) or are electrochromic under electrolysis potentials;<sup>156</sup> as so, the optimal thickness of a uniform catalyst overlayer should be of few nanometres. Alternatively, catalysts can be applied as random or regular arrays with low geometric filling fractions, which reduces the negative effect of opacity.

The application of co-catalysts or other surface treatments plays a crucial role in the stability of the photoelectrode in contact with an electrolyte solution, a process that is known to be the competition between anodic/cathodic dissociation and redox reaction, controlled by thermodynamic and kinetic parameters, respectively.<sup>147,157</sup> The selection of a suitable co-catalyst and surface treatment requires a set of experimental studies to conclude on the products of HER/OER photo(corrosion) and other secondary reactions.<sup>155</sup> This issue was reviewed in detail by Ros *et al.* The authors concluded that the application of thin metallic films with a few nanometers thick (enough to be transparent) that could act both as catalysts and as a protective layer, was not an effective solution.<sup>63</sup> Indeed, for protecting short bandgap photoabsorbers, such as Si, CZTS or similar materials, protective layers of several tens of nanometers thick are required. For that, TiO<sub>2</sub> and NiO have been the most studied materials, the former acting as a HER/OER catalyst and the later as an OER catalyst. Still, the authors highlight that for both TiO<sub>2</sub> and NiO, the deposition technique, and the intermittent polarization during long-term operation can hinder their electronic structure, therefore compromising stability.

**2.1.4. Electronic conductive substrates.** The preparation of PEs for practical PEC applications implies the use of suitable substrates to have a rigid or semi-rigid component that can be easily inserted or removed from the reactor. Given the operational constraints of PEC water splitting, and depending on the arrangement of electrodes, the selection of the substrate can play an impactful role in the overall performance of a PEC device. The most simple and effective substrate for electrolysis is a metallic substrate, such as those used in alkaline electrolysis. However, for PEC applications, metallic substrates present critical disadvantages, such as opacity, prohibiting their use in back-illuminated arrangements and in tandem applications where sunlight should pass from one sub-cell to the other. Also, metallic substrates often require expensive solutions to assure durability and resistance to corrosive environments.<sup>158</sup> Still, different examples of their successful use for PEC applications can be found in the literature. Lv *et al.* reported porous iron oxide-based photoanodes prepared by modulated anodization.<sup>159</sup> Continuous square-wave modulation and negative potentials (−2 V to −10 V range) at 100 Hz allowed the etching anodization of the metallic Fe foil and incorporation of Sn-dopant. These iron oxide electrodes exhibited a stable performance over 10 h, yielding *ca.* 0.3 mA cm<sup>-2</sup>. Dias *et al.* compared the performance of WO<sub>3</sub> prepared over TCO-glass substrates and over tungsten foils and concluded that the intrinsic resistivity of the TCO-glass substrate limited the photocurrent plateau, which was not observed in PEs prepared over metal foils.<sup>160</sup> Also, during



a 72 h aging test, no degradation was observed in WO<sub>3</sub> PEs prepared on metal substrates, yielding *ca.* 1.5 mA cm<sup>-2</sup>.

Despite some performance limitations, TCO-glass remains the most used substrate for solar water-splitting electrodes. TCOs present unique advantages, such as (i) a combination of optical transparency with electrical conductivity; (ii) tunable energy bandgap (2.8–4.2 eV); and (iii) chemical and mechanical stability.<sup>161,162</sup> As a result, TCOs have the advantages of metals typically used for electrical contacts, while avoiding their main drawbacks, *i.e.* opacity and corrosion issues.<sup>163,164</sup> Among TCOs, ITO is the most broadly studied and is used in multiple optoelectronic applications. However, ITO lacks temperature and chemical stability, which compromises several preparation steps of the PE downstream, and since indium is a scarce and toxic element, its use for PEC devices has been limited.<sup>165</sup> FTO-coated glass substrates became the best alternative to ITO for PEC applications, presenting a wide energy bandgap (*ca.* 3.6 eV), low-cost production, thermal stability, chemical inertness, low sheet resistance and high optical transmittance (usually series sheet resistance  $\leq 15 \Omega \square^{-1}$  and transmittance  $\geq 80\%$  at 550 nm).<sup>166,167</sup> Also, FTO thin films can be easily prepared by simple and low-cost methods, such as screen printing, doctor blade, and spray pyrolysis.<sup>168–171</sup> Due to their versatility, the preparation of FTO glass substrates by spray pyrolysis allows for full optimization and customization for PEC applications. By varying the sprayed volume and the [F]/[Sn] atomic ratio and using a figure of merit to evaluate the optoelectronic performance of FTO substrates, Pinheiro *et al.* were able to tune the optoelectronic properties of these substrates for PV applications, achieving PCE improvements of 8.9% in DSSCs and 4.7% in PSCs.<sup>172</sup> Hence, when upscaling of PEs is foreseen, developing custom-made large-area FTO-glass substrates is essential to optimize the optoelectronic performance of the resulting PEs.

Although FTO-glass substrates have remained the preferable substrate for the preparation of PEC photoelectrodes, they also present non-negligible limitations. First, when practical mounting issues arise, such as screwing, sealing, and sudden temperature variations (especially when the PE is used simultaneously as a PEC cell window), the risk of cracking becomes high. Furthermore, the use of rigid substrates greatly limits the use of roll-to-roll manufacturing techniques, which have been widely used in large-scale production processes. Aiming to overcome these limitations, some studies reported the preparation of flexible PEs for solar water splitting, *i.e.*, the deposition of semiconductor materials over flexible substrates. Hou *et al.* reported an innovative ZnO-based flexible PE prepared on micropatterned polyurethane acrylate (PUA) fabricated by a mold transfer process.<sup>173</sup> A thin layer of ITO was deposited on the polyurethane acrylate substrate to assure electrical conductivity, and subsequently, one dimensional ZnO nanorods (NRs) arrays were prepared by a facile seed layer and hydrothermal method. These PEs were able to generate 0.52 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>. In a different approach, Thakur *et al.* demonstrated a flexible photoanode prepared on micro-porous laboratory filter paper decorated with polypyrrole, which displayed a bandgap of 1.98 eV, yielding 9.5 mA cm<sup>-2</sup> at 1.23 V<sub>RHE</sub>.<sup>174</sup>

## 2.2. Counter-electrodes

The classic configuration of a PEC device (Fig. 3(a)), where only one of the electrodes is photoactive, comprises a semiconductor operated as a working electrode (WE) and a “dark electrode”, the so-called counter-electrode (CE), where the complementary HER or OER takes place.<sup>175</sup> As so, the CE plays an equally relevant role in the overall STH efficiency, and an inappropriate selection of materials or design criteria can significantly hinder the performance of a PEC device, as it must address some key features: (i) high surface area, at least the double than that of the PE, and good catalytic activity (to prevent undesirable photocurrent limitations originated by the overpotentials at the counter-electrode); (ii) good stability on strong acidic or alkaline media; (iii) ductility for enabling different assembling strategies, avoiding inhomogeneous pathways of charge carriers and parasitic ionic paths in the electrolyte; and finally (iv) non-toxicity, abundance and low-cost. Unfortunately, there are few available materials that meet all the aforementioned requirements, which makes some noble metals the obvious choice for lab-scale experiments. As a result, Pt has been used as a CE in small PEC devices for decades, mainly due to its low overpotential (*ca.* 0.1 V) and high chemical stability over a wide range of electrolytes and pH values. Almost like a standardized procedure, using a spiral-shaped Pt next to the PE became the most reported configuration among PEC characterization setups. Despite being commercially available in multiple configurations, the price and scarceness of Pt make it non-viable for large-area applications.<sup>176</sup> Platinized meshes appeared as an alternative for larger devices, usually consisting of Pt-coated titanium or niobium meshes, having a higher surface area, while preserving the stability and low overpotential features.<sup>39,43</sup> Even so, the amount of Pt incorporated in these electrodes, as well as other critical raw materials (CRMs), like Ti or Nb, make them suitable only for pilot-scale demonstration devices.

Research efforts have been made to develop highly efficient and stable non-noble transition-metal-based and metal-free electrodes for water splitting applications. Liu *et al.* proposed an efficient flexible binder-free electrode for the HER, showing a low overpotential, compared with Pt, and notable electrocatalytic performance.<sup>177</sup> The authors reported the electrophoretic-deposition of MoSe<sub>2</sub> nanosheets on carbon cloth, which is a low-cost textile with high electric conductivity, large surface area and excellent flexibility and strength, which facilitates its integration in PEC devices. The prepared electrodes show a large number of exposed active edges due to the nanostructured MoSe<sub>2</sub> sheets and the large surface area of the 3D-carbon cloth support, displaying remarkable catalytic activity. Binder-free catalysts to be used in the counter-electrode half-reaction own clear advantages as the catalyst is solely formed by the aggregation or self-supporting nature of the active material, improving: (i) activity, as the absence of binders allows the reactants to reach the active site easily, (ii) stability, as some binders can degrade or react with the active components of the catalyst under harsh operating conditions, (iii) mass transport within the catalyst structure, and (iv) selectivity towards the desired products. However, some binder-free catalysts/counter-electrodes may have a fragile



structure, being more susceptible to physical damage, which can sometimes make it difficult to shape counter-electrodes to specific reactor designs and catalyst recovery. However, for a given catalytic process, binders may provide additional stability or tailor the catalyst properties.<sup>178</sup> Similar approaches are being pursued using half-membrane electrode assemblies (H-MEA), where the separate evolution of O<sub>2</sub> and H<sub>2</sub> gases is assured by bonding the CE to an ion-exchange membrane. When MEAs are used, *i.e.* catalyst-coated membranes (CCM), the direct coating can be made on commercially available membranes (*i.e.* Aemion, Sustainion, Piperion, Nafion) or substrates, *e.g.* carbon cloth or carbon paper. Usually, a catalyst-ink is applied by spray-coating or doctor-blading, followed by thermal and pressure-finishing, with subsequent characterization, for assessing mechanical stability, morphology, and conductivity.<sup>179</sup> MEAs can be prepared with nanoparticles of non-noble metals, such as Ni-nanoparticles, sprayed onto a hydrophilic carbon-based current collector particle support, which is normally bonded to an ion-exchange membrane using an ionomer glue. The shape of the electrode can be adapted to any reactor design and has the advantage of enabling the operation with a vapour phase on the HER side. An example of a carbon-based electrode was reported by Wang *et al.*,<sup>180</sup> where 3-D carbon paper/tubes/cobalt-sulphide sheets were fabricated and tested as electrodes for the HER and OER. These electrodes presented high activity and stability for both cases, which was attributed to the increased exposure of active sites, improved electron transport, and good release of gaseous products, with demonstrated continuous water electrolysis over 2 h, presenting an overpotential of only 64 mV comparatively to Pt/C electrodes. Noble-metal free transition metal chalcogenide catalyst systems, *i.e.* (Fe,Ni)<sub>9</sub>X<sub>8</sub> and NiX with X = S, Se, Te, and pentlandites (Fe<sub>4.5</sub>Ni<sub>4.5</sub>S<sub>8</sub>), have shown quite promising results when used as CEs in water electrolysis systems.<sup>181,182</sup> Solution-based bottom-up approaches (precipitation, solvothermal) can be implemented for the synthesis of small-particle size seleno- and telluro-pentlandites. Still, the variation of chalcogenide/pentlandite composition as well as particle size distribution should be optimized for each PEC system. Fig. 5 shows schematically the incorporation of an MEA in two different PEC device architectures.

### 2.3. Anode/cathode separation materials and designs

Typical PEC devices require the incorporation of an internal separator to prevent gas mixing, avoiding the need for further processing downstream and assuring safer and more efficient operation. In current PEC systems, and considering the operation at  $\leq 10 \text{ mA cm}^{-2}$ , the membrane can represent up to 17% of the total cost of the device, more than that of the electrodes.<sup>38</sup> Indeed, as demonstrated by Grimm *et al.*, the membrane is the most expensive component of a PEC module.<sup>183</sup> Ion-exchange membranes, *i.e.* cation or anion exchange membranes (CEMs, AEMs), are common choices, being also widely applied in electrolyzers, fuel cells and redox flow batteries.<sup>184–186</sup> Whatever the choice for the membrane, inserting a selective porous separator in a PEC cell represents an additional resistance to species transport between electrodes due to its intrinsic properties (conductivity and permeability to the evolved gases) and due

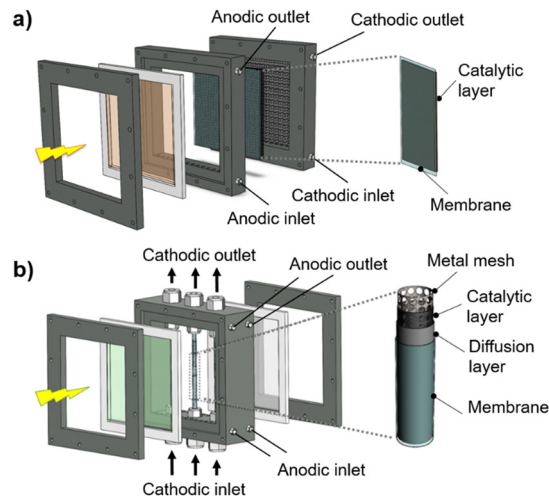


Fig. 5 Schematic representation of MEAs comprising catalyst-coated membranes assembled in different PEC device architectures: (a) back-to-back configuration; and (b) side-by-side configuration.

to its influence on the device design, *i.e.* the location of the membrane, in a large-area device, influences directly the distance between the electrodes. As an example, Vilanova *et al.* demonstrated that increasing the distance between two Pt-Ti meshes from 1 cm to 9 cm resulted in an increase of the ohmic losses from 73 mV to 297 mV, at a current density of  $8.1 \text{ mA cm}^{-2}$  (simulating 12% STH efficiency) in 1.0 M KOH.<sup>43</sup> In another work, Berger *et al.* concluded that, among the main properties of membranes, the permeability to gases is the one that has the greatest influence on the device configuration.<sup>187</sup> Apart from safety-related problems, gas crossover in a PEC device leads to an obvious decrease in net H<sub>2</sub> collection, therefore requiring the operation at higher current densities to collect the same amount of H<sub>2</sub>. Finally, depending on the arrangement of PES, namely in tandem configurations, membranes may need to address other critical requirements, such as high transmittance to sunlight.

Considering water electrolysis in acidic media, perfluorinated sulfonic acid (PFSA) polymers, such as DuPont's Nafion, are the most commonly used.<sup>188</sup> Despite being commercially available for large-area applications, CEMs still have a high cost per unit area ( $0.25\text{--}0.85 \text{ USD cm}^{-2}$ ). Similar to CEMs, AEMs comprise fixed positive charges, which are compensated by mobile anions, with costs ranging from  $0.16 \text{ USD cm}^{-2}$  to  $2.5 \text{ USD cm}^{-2}$ . Typical AEMs include reinforced PET or PEEK, which have been reported in several PEC devices. Poly(norbornene)-based resin membranes are among the most expensive; they present an innovative ionomer structure that contains a poly(norbornene) backbone with quaternary ammonium functional groups, with an integrated reinforcement layer that provides enhanced mechanical properties.

In contrast to conventional monopolar membranes, bipolar membranes (BPMs) were not designed for ion transport across them. These polymeric membranes comprise a negatively charged cation-exchange layer and a positively charged anion-exchange layer. BPMs promote the electro-dissociation of water into protons and hydroxide ions at the bipolar junction, without





any gas formation; still, their use in PEC devices has been barely assessed.<sup>189</sup> Even so, BPMs can be an interesting approach for PEC devices that comprise photocathodes and photoanodes, allowing them to pair two different electrolytes suitable for their respective oxidation and reduction reactions. Moreover, using strong acidic and basic electrolytes at the cathode and anode sides, respectively, reduces the overall losses in a BPM-based cell and the induced pH gradient may tune the redox potentials for a better alignment of the PE band edges. In this arrangement, H<sup>+</sup> ions can pass through the cation exchange layer, while OH<sup>-</sup> can pass through the anion exchange layer; the quantity of generated protons and hydroxides at the BPM matches the consumption at the electrodes, maintaining a constant pH difference across the membrane. The total bias over the cell is not increased nor decreased by the water dissociation in the BPM, since the potential difference required to dissociate water over the membrane causes an equal shift in redox potentials for the HER and OER, due to the pH difference between electrodes. Vermaas *et al.* demonstrated for the first time photoassisted water splitting performed in a PEC device with an acidic cathodic compartment and a neutral or alkaline anodic compartment separated by a BPM.<sup>190</sup> The use of a BPM allowed for maintaining the pH gradients and reducing the needed bias potential. PEC water splitting was promoted by a Pt cathode immersed in 1 M H<sub>2</sub>SO<sub>4</sub> (pH 0), and by a BiVO<sub>4</sub> photoanode, immersed in 1 M KOH (pH 14), resulting in a current density of 2 mA cm<sup>-2</sup> at 1.3 V; the authors reported that a minor fraction of other ions was transported through the membrane due to imperfect selectivity.

Aiming at eliminating the resistance imposed by selective internal separator membrane-free PEC devices and integrated PEC (IPEC) systems have been proposed and demonstrated. However, these approaches often result in higher ohmic losses, due to the larger electrode spacing, or in additional and unnecessary complexity, due to the need of placing extra parts or integrating convoluted electrolyte paths to prevent gas mixing. Nevertheless, for small-scale applications, such as standard lab measurements, membrane-less devices are widely used, such as the Cappuccino PEC cell.<sup>191</sup> Here, given the small illuminated area (*ca.* 0.5 cm<sup>2</sup>), the amount of evolved gases does not represent any safety concern, though the energy performance becomes affected.<sup>175</sup> Landman *et al.* proposed an innovative membrane-free PEC design where H<sub>2</sub> and O<sub>2</sub> are generated in separated cells, and a solid-state redox system mediates the ion exchange between the anode and cathode, both immersed in alkaline aqueous solutions. The auxiliary electrodes (AEs) were a NiOOH/Ni(OH)<sub>2</sub> redox couple, which had the possibility of being cycled multiple times; electrons are transferred between these AEs through metal wires.<sup>192</sup> This approach presents some advantages over the typical approach that comprises a physical gas separator: (i) H<sub>2</sub> can be produced in a centralized generator that can be placed at the end-user's location; (ii) gas mixing inside the PEC reactor is totally eliminated; and (iii) the O<sub>2</sub> evolution compartment does not need to be sealed, thus avoiding piping costs. Still, for large-area applications, membrane-free systems are far from implementation, mainly because the auxiliary

electrodes require frequent replacement and the whole system needs further experimental validation.

### 3. Upscaling of PEC water splitting

Despite being investigated for over 50 years, PEC-H<sub>2</sub> production is still far from commercialization, with few demonstrated large-area devices. Although remarkable progress has been made in materials for water splitting, research work is still mostly focused on lab-scale experiments.<sup>193–196</sup> Nevertheless, for PEC water splitting to become a competitive green-H<sub>2</sub> production technology, it is required that research devices are designed to be scalable. Moreover, it is being recognized that the PEC device design plays an equally important role as the photoelectrode during the upscaling process. Device architectures with an illuminated area of at least 50 cm<sup>2</sup> can be considered as a first step towards large-scale applications, given the limitations imposed by some manufacturing equipment and by the electronic resistance of some substrates, namely transparent conductive oxides (TCOs). Indeed, some authors who have addressed the challenge of upscaling photoelectrodes (PEs) for PEC water splitting have considered 50 cm<sup>2</sup> as a benchmark for labelling their devices as “large-area”.<sup>43,79,197,198</sup> Alternatively, scalability can be achieved by coupling module units of smaller dimensions. Therefore, materials engineering and fabrication techniques must be cleverly optimized envisaging upscaling, while preserving the performance of state-of-the-art materials and assuring a low-cost assembly towards higher TRL.

The following subsections address relevant strategies and fabrication techniques for the preparation of efficient large-area PEs, as well as PEC device architectures suitable for upscaling.

#### 3.1. Engineering solutions for large-scale photoelectrodes

The upscaling of efficient PEs remains one of the main challenges for the industrial implementation of PEC-H<sub>2</sub> technology. Although nanostructured materials have been greatly optimized at the lab-scale (usually ≤ 5 cm<sup>2</sup>),<sup>67,140</sup> upscaling top-performing PEs towards the expected dimensions of commercial PEC panels (≥ 1 m<sup>2</sup>) without significant losses is a demanding task.<sup>199,200</sup> Indeed, Hankin *et al.*<sup>201</sup> demonstrated that the performance of PEC devices is greatly affected by the geometry of the electrodes, especially for PEs with the smallest dimension > 0.1 m. Table 1 summarizes common challenges during the upscaling of PEs and provides a comparative overview of the advantages and challenges of using smaller and larger PEs in PEC devices.

As mentioned before, FTO-glass stands as the most widely used PE substrate in PEC water splitting.<sup>202,203</sup> These substrates are resistant to corrosion by commonly used electrolyte solutions and allow pairing of two photoabsorbers in series, which is a requirement for PEC–PV tandem systems, often considered the most feasible arrangement for reaching the STH efficiency targets.<sup>77,195</sup> However, compared to metallic substrates, glass-coated substrates have a relatively high sheet resistance (usually ranging from 7–15 Ω □<sup>-1</sup>). Some promising strategies for overcoming this drawback have been demonstrated over the

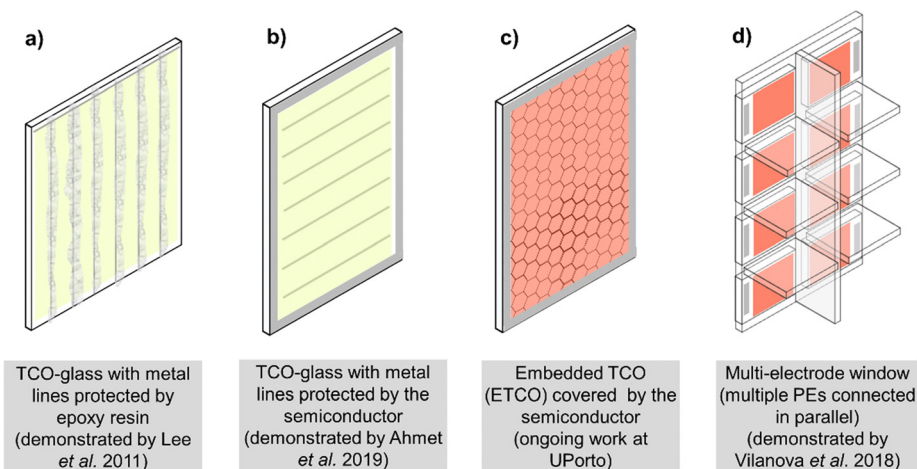


**Table 1** Comparative overview of the main challenges during upscaling of PEs and the corresponding PEC devices;  – easy to implement/achieve;  possible to achieve, but with severe constraints/limitations;  – very difficult to implement/achieve

	Lab-scale ( $\leq 5 \text{ cm}^2$ )	Multi-PE ( $\geq 25 \text{ cm}^2$ )	Large PEs ( $\geq 50 \text{ cm}^2$ )	PEs with metal lines ( $\geq 50 \text{ cm}^2$ )	Modular device ( $\geq 100 \text{ cm}^2$ )	PEC panel ( $\geq 1 \text{ m}^2$ )
Schematics of the PE(s)						
Preparation	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>
Testing	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>
Efficiency	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>
Stability	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>
Active area	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>	<input checked="" type="checkbox"/>
Published device for demonstration (example)	 Cappuccino Cell <sup>44</sup>	 CoolPEEC cell <sup>206</sup>	 CoolPEEC cell <sup>43</sup>	 PortoCell <sup>39</sup>	 CoolPEEC module <sup>99</sup>	 Artiphycion <sup>199</sup>

past decade – Fig. 6. The use of conductive lines across the substrate seems to be an obvious and easy approach for reducing the electronic path until the current collection point.<sup>204</sup> Lee *et al.* successfully demonstrated a  $131 \text{ cm}^2$   $\text{WO}_3$  PE prepared on an FTO glass substrate with an embedded silver grid – Fig. 6(a); this PE remains the largest prepared on a glass substrate containing metal lines.<sup>100</sup> Silver current collector grid lines (width  $\times$  height:  $0.6 \text{ mm} \times 10 \mu\text{m}$ ) were prepared by screen-printing a silver paste on the FTO substrates, followed by annealing at  $180 \text{ }^\circ\text{C}$ . This substrate presented a resistance of *ca.*  $1 \Omega$  between two points

$5 \text{ cm}$  apart, considerably lower than the standard specification of the substrate, which is  $10 \Omega$  for the same distance ( $10 \Omega \square^{-1}$ ).  $\text{WO}_3$  was later deposited between the silver lines by screen printing and sintered at  $500 \text{ }^\circ\text{C}$ . Finally, the silver lines were encapsulated with a non-conductive and non-corrosive epoxy resin. This  $131 \text{ cm}^2$  PE generated a current density of  $1.18 \text{ mA cm}^{-2}$  at  $1.23 V_{\text{RHE}}$ . In 2019, Ahmet *et al.* reported a  $50 \text{ cm}^2$   $\text{BiVO}_4$  PE prepared on an FTO glass with electrodeposited Ni lines that allowed achieving a current density of  $1.5 \text{ mA cm}^{-2}$  at  $1.23 V_{\text{RHE}}$  – Fig. 6(b).<sup>79</sup> This work followed a different approach



**Fig. 6** Demonstrated strategies for overcoming the ohmic losses in large FTO glass substrates: (a) metal lines deposited on TCO glass substrates protected by epoxy resin; (b) metal lines deposited on TCO glass substrates and protected by the semiconductor, applied directly over the metal lines; (c) metal lines applied on ablated glass, covered by TCO, prior to the deposition of the semiconductor; (d) multi-PE, comprising several small-size PEs externally connected in parallel.



to applying metal lines in a large-area glass substrate; the photoactive material was deposited over the whole substrate, including exposed FTO and Ni lines. These lines (7 cm long, 2 mm wide) were electrochemically deposited in a pre-treated and masked FTO glass, followed by the deposition of a 10 nm thick tin oxide (SnO<sub>2</sub>) layer and the 200 nm thick BiVO<sub>4</sub> layer.

Despite the current density generated by these large-area PEs (Fig. 6(a) and (b)) was less than half of the current density generated by equivalent small-sized PEs, the use of metal lines demonstrated to be an effective approach to overcome the electronic resistance across the substrate. However, the stability of these metal lines remains a massive challenge, since they need to be protected against corrosion, which is hard to achieve by encapsulation, such as using an epoxy resin, or covering the substrate with photoactive materials.<sup>21,205</sup> Some innovative approaches have been proposed for overcoming most of the limitations imposed by the application of metal lines in conductive glass substrates. The ETCO (embedded TCO) – Fig. 6(c) – basically consists of applying metal lines beneath the TCO layer, inside ablated channels. This approach was reported by Ivanou *et al.*<sup>206</sup> for a 6.5 × 6.5 cm<sup>2</sup> FTO glass substrate. ETCO presented full stability and suppressive behaviour for recombination in DSSCs, allowing an increase in the cross-sectional area of current collectors and enabling the use of metals with good specific conductivity. However, this concept still lacks a further demonstration for PEC applications, *i.e.* with recirculation of corrosive electrolytes under uninterrupted solar illumination. The use of metal lines in large-area PEs, with its associated problems, can be overcome by simply connecting small PEs in parallel – Fig. 6(d), fulfilling large-area requirements while avoiding corrosion-related problems, as demonstrated by Vilanova *et al.*<sup>207</sup> The authors developed a multi-electrode support with 5 × 10 cm<sup>2</sup> that can be assembled as a PEC cell window. Through the connection of eight PEs in

parallel using this support, a slightly higher photocurrent density and photopotential were reported, compared with a single 50 cm<sup>2</sup> α-Fe<sub>2</sub>O<sub>3</sub> PE. Apart from the significant performance losses imposed by the substrates, the successful demonstration of upscaled PEs is also related to the preparation technique of the semiconductor. Given the obstacles presented by upscaling of PEs, “classic” metal oxide PEs have been the subject of relevant studies aiming at increasing the active area without significant losses in performance while improving their stability under standard operating conditions. Table 2 provides an overview of the current status of large-area PEC water splitting.

### 3.2. Brief overview of scalable fabrication techniques

Deposition techniques can be divided into chemical and physical, as well as into dry and wet. Dry techniques include processes such as thermal and plasma-assisted atomic layer deposition (ALD), sputtering and evaporation, while wet techniques include the application of inks/pastes, solution-based methods or chemical sprays with subsequent thermal processing.<sup>212</sup> Most lab equipment is optimized and calibrated to prepare highly efficient PEs only for small-area applications,<sup>156,213</sup> record-performing PEs usually have few square centimetres.<sup>39,175</sup> Moreover, even when the preparation techniques allow the use of large-area substrates, it is difficult to ensure uniform layers across the entire PE area. Ahmet *et al.* assessed this problem by dividing their 50 cm<sup>2</sup> BiVO<sub>4</sub> PE into four equal parcels of 12.5 cm<sup>2</sup>,<sup>79</sup> each of them was tested individually. The authors concluded that the variation between the samples remained below 13%. However, even when the prepared material is uniform, the performance deviation between small and large samples is often attributed to the preparation technique. As an example, Lee *et al.* reported that the shape of the metal grid applied on the 131 cm<sup>2</sup> WO<sub>3</sub> PE did not allow using the 5 multi-step deposition procedure

**Table 2** Overview of demonstrated PEs for PEC water splitting with an active area ≥ 25 cm<sup>2</sup>, ordered by active area; Type I corresponds to the wired back-to-back design, Type II corresponds to the wireless back-to-back design, Type III corresponds to the side-by-side design, and Type IV corresponds to the membrane-less dual reactor device design; adapted from ref. 97

Top absorber [bottom or side absorber]	Electrolyte/pH	Area (cm <sup>2</sup> )	Current density (mA cm <sup>-2</sup> )/η <sub>STH</sub> (%)	Stability (h)	PEC architecture	Year	Ref.
FTO/WO <sub>3</sub> /Mo–BiVO <sub>4</sub> /Co–Pi	0.5 M Na <sub>2</sub> SO <sub>4</sub> + 0.1 M KH <sub>2</sub> PO <sub>4</sub> /pH 7	25	2.20@1.23 V	1	Type I	2018	Yao <i>et al.</i> <sup>208</sup>
FTO/Fe <sub>2</sub> O <sub>3</sub>	1.0 M KOH/pH 13.6	26	1.11@1.45 V	—	Type III	2018	Vilanova <i>et al.</i> <sup>207</sup>
LaTiO <sub>2</sub> N	0.1 M Na <sub>2</sub> SO <sub>4</sub> + 1 M NaOH/pH 13.4	40	0.56@1.23 V	—	Type I	2019	Dilger <i>et al.</i> <sup>209</sup>
W/WO <sub>3</sub>	3 M MSA acid/pH 0.14	49	0.90@1.45 V	—	Type I	2014	Lopes <i>et al.</i> <sup>39</sup>
FTO/Fe <sub>2</sub> O <sub>3</sub> [2 × SHJ Si]	1.0 M KOH/pH 13.6	50	1.05@1.45 V	1008	Type III	2017	Vilanova <i>et al.</i> <sup>43</sup>
FTO/Ag–Pt/W:BiVO <sub>4</sub> /CoPi [2 × SHJ Si]	0.1 M KPi buffer/pH 7	50	1.50@1.23 V/ 2.1	—	Type III	2019	Ahmet <i>et al.</i> <sup>79</sup>
FTO/Cu grids/Cu <sub>2</sub> O/AZO/Ga <sub>2</sub> O <sub>3</sub> /RuOx	0.5 M Na <sub>2</sub> SO <sub>4</sub> + 0.1 M KPi/pH 5	50.4	3.70@0 V	—	Type I	2016	PECDEMO <sup>102</sup>
FTO/Fe <sub>2</sub> O <sub>3</sub>	1.0 M KOH/pH 13.6	51	0.47@1.45 V	—	Type I	2014	Lopes <i>et al.</i> <sup>39</sup>
FTO/Au grids/NiFe–BiVO <sub>4</sub>	0.1 M H <sub>3</sub> BO <sub>3</sub> + Na <sub>2</sub> B <sub>4</sub> O <sub>7</sub> ·10H <sub>2</sub> O/pH 8.4	54.3	2.55@1.23 V	5	Type I	2020	Huang <i>et al.</i> <sup>210</sup>
2 series-connected a-Si:H/μc-Si:H tandem PV cells	1.0 M KOH/pH 13.6	64	—/3.9	40	Type II	2016	Turan <i>et al.</i> <sup>21</sup>
FTO/Fe <sub>2</sub> O <sub>3</sub> [c-Si]	1.0 M KOH/pH 13.6	100	0.56@1.84 V/0.71	8	Type IV	2020	Landman <i>et al.</i> <sup>211</sup>
FTO/WO <sub>3</sub> [2 × SHJ Si]	0.5 M H <sub>2</sub> SO <sub>4</sub> /pH 0.3	131	1.18@1.23 V/ 2	—	Type II	2011	Lee <i>et al.</i> <sup>100</sup>
FTO/Ti–Fe <sub>2</sub> O <sub>3</sub>	1.0 M KOH/pH 13.6	200	2.00@1.45 V	48	Type III	2019	Vilanova <i>et al.</i> <sup>97</sup>
Mo–BiVO <sub>4</sub> /CoPi [Si PV]	0.1 M KPi buffer/pH 7	16 000	—	—	Type I	2017	Tolod <i>et al.</i> <sup>199</sup>





**Table 3** Summary of preparation techniques suitable for PEC water splitting materials (adapted from ref. 42); ○ ○ ○ corresponds to “Low” and denotes minimal impact or resource consumption, while ● ● ● corresponds to “High” and denotes substantial impact or resource consumption

Synthesis	Techniques	Characteristics				
		Waste	Time-consuming	Quality	Cost	Comments
Physical	Spin coating (wet)	● ● ○	● ○ ○	● ○ ○	○ ○ ○	Limited for multilayers
	Evaporation	○ ○ ○	● ● ○	● ○ ○	● ● ○	Low melting point only
	Magnetron sputtering	○ ○ ○	● ● ●	● ● ●	● ● ●	Expensive equipment
Chemical	ALD	○ ○ ○	● ● ●	● ● ●	● ● ●	Limited chemistries; expensive equipment
	CVD	● ● ●	● ● ○	● ○ ○	● ● ○	High contamination
	Plasma ALD/CVD	○ ○ ○	● ● ○	● ● ●	● ● ●	Expensive equipment
	Spray pyrolysis (wet)	○ ○ ○	● ● ○	● ● ○	● ○ ○	Ideal for oxides
	Ink jet (wet)	● ○ ○	○ ○ ○	● ○ ○	● ○ ○	Ideal for patterned areas
	Electro/electroless plating (wet)	● ● ●	● ○ ○	● ○ ○	○ ○ ○	Ideal for metals
	Sol-gel/slurry (wet)	● ○ ○	● ○ ○	○ ○ ○	○ ○ ○	Low reproducibility
Solution-based methods (hydrothermal)	● ○ ○	● ● ○	● ● ○	○ ○ ○	Low-cost, limited quality control	

optimized for smaller samples, which do not have a metal grid.<sup>100</sup> Consequently, instead of presenting an optimal thickness of 25 μm, large WO<sub>3</sub> PEs were prepared *via* a one-step deposition procedure, presenting a thickness of 5 μm. Table 3 presents a summary of comparison between technologies suitable for preparing large-area PEs.<sup>42</sup>

When a preferable preparation technique does not allow achieving a uniform semiconductor deposition across the substrate, having a multi-PE window could be an interesting approach, as it allows working with small PEs prepared using state-of-the-art techniques/equipment. Even so, this multi-electrode window needs further optimization, especially due to the increased complexity brought by electrical connections and the associated costs.<sup>207</sup> As described by the authors, this window limits the implementation of tandem arrangements, since the electrical connections made at the back of the cell block part of the incident light. The research group of Lee has achieved a significant breakthrough in large-scale PE manufacturing using inkjet printing.<sup>214</sup> The proof-of-concept of fully inkjet-printed α-Fe<sub>2</sub>O<sub>3</sub> layers on an FTO glass substrate was demonstrated with a 20 × 25 array and photo-illumination area of 500 cm<sup>2</sup>; the authors also reported the formulations for preparing other semiconductors. However, they did not provide any information regarding the performance of large-area inkjet-printed PEs or on economic and environmental analyses.

### 3.3. Device architectures for practical applications

Among the various architectures used in PEC devices, two main groups of planar architectures can be identified, based on the implemented strategy for separating the evolved H<sub>2</sub> and O<sub>2</sub>: (i) PEC reactors comprising an internal mechanism to separate the evolved gases; and (ii) membrane-free systems comprising separated cells.<sup>42</sup> Considering the strategy for connecting the electrodes, it is possible to divide planar PEC device architectures into four main types: the back-to-back design, where the electrodes can be connected wired or wireless – type I and type II, respectively; the side-by-side design – type III; and the membrane-less dual reactor device – type IV. Types I, II and III use a membrane separator for dividing the anode from the cathode inserted in a single container. Fig. 7 shows schematically these four main PEC device architectures. Table 4 presents

more detailed information regarding PEC reactors of each type, designed and implemented for large-scale solar H<sub>2</sub> products. The back-to-back PEC design containing the electrodes in a wired configuration (type I) is the most studied architecture for both lab- and large-scale applications. This design is quite similar to the well-known alkaline electrolyser (AEC), comprising a reservoir for the electrolyte wherein the electrodes are immersed and physically separated by an ion-exchange membrane. Over the past decade, different approaches were reported for scaling up this family of devices: (i) in 2011, a 100 cm<sup>2</sup> “stacked” PEC reactor was reported for the first time by the Imperial College of London;<sup>215,216</sup> (ii) in 2014 UPORTO developed the 100 cm<sup>2</sup> “PortoCell”, which was further improved in 2016;<sup>39,217</sup> (iii) in 2015, the European Union (EU) project ArtipHyction developed the largest PEC prototype to date, with an irradiated surface area of 1.6 m<sup>2</sup>;<sup>218</sup> and (iv) in 2019, EPFL patented an integrated PEC (IPEC) device, whose concept was validated experimentally using a close electronic integration between the III–V-based photoabsorber (4 cm<sup>2</sup>) and the IrRuOx-Pt-based electrocatalyst.<sup>17</sup> Holmes-Gentle *et al.* reported an STH efficiency of >20% at a H<sub>2</sub> production rate of >2.0 kW (>0.8 g min<sup>-1</sup>),<sup>219</sup> landmark results with PV-grade PEs, such as Si-perovskite cells. Although great efforts have been pursued to successfully upscale type I PEC cells, the arrangement of electrodes for a PEC–PV tandem configuration has always been a limiting factor for the efficient utilization of the solar spectrum, and also required an optimized electrolyte flow pattern for efficient dissipation of heat and evolved gases. The upgraded version of the “PortoCell” was the first PEC design that included an optimized flow pattern and simultaneously addressed ohmic losses across the electrolyte reservoir; still, this design presents some disadvantages under real operating conditions.<sup>217</sup> On the other hand, the ArtipHyction prototype was the first device that validated the transfer from lab-scale cells to pilot-scale units; however, its viability was compromised by the low overall PEC performance: STH conversion efficiency of *ca.* 2% with a 5% decrease over 1000 h of continuous operation.<sup>218</sup>

Type II PEC reactors are rarely reported; an example of this architecture was successfully demonstrated in 2015 by Walczak and his co-authors<sup>224</sup> with the development of the so-called Louvered cell. In the Louvered prototype, planar macroscale



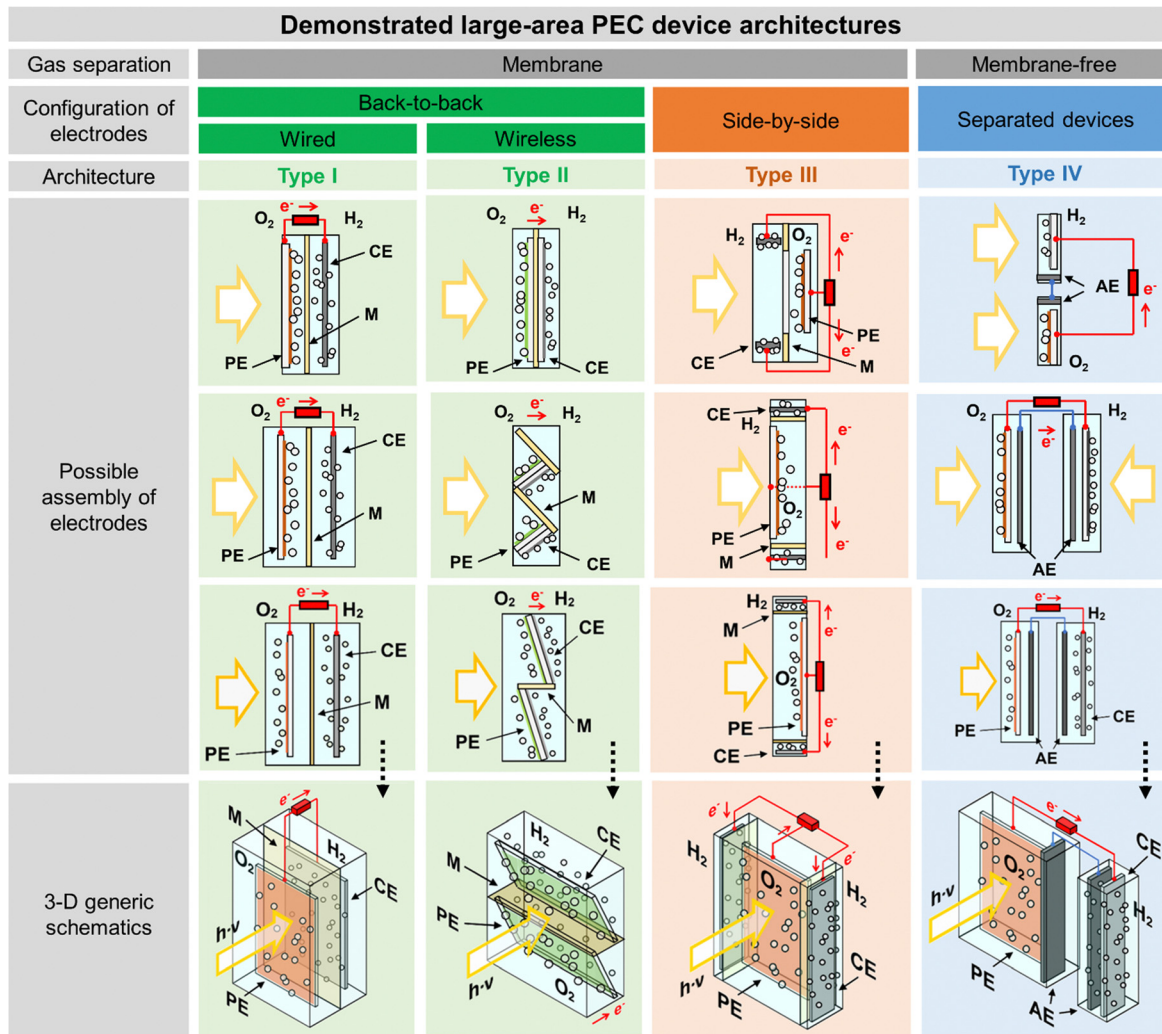


Fig. 7 Schematic representation of the four main planar PEC device architectures. PE – photoelectrode; CE – counter-electrode; M – membrane; AE – auxiliary electrode.

tandem PEs are installed at a certain angle in a chamber; the PE compartments are separated by an anion-exchange membrane.<sup>224,225</sup> During the water splitting process, O<sub>2</sub> and H<sub>2</sub> are collected in different compartments of the cell, respectively – cf. Fig. 7.<sup>224,225</sup> The overall efficiency of these PEC reactors is highly dependent on the performance of the tandem PEs but is equally important to ensure a proper ionic transport with low overpotentials.<sup>228</sup>

Within the type II architecture, integrated tandem monolithic PEs, where the top and bottom absorbers are combined into a single monolithic electrode, offer several practical benefits. Since the illumination path is confined through a single thin monolithic device, optical scattering, and parasitic absorption losses are mitigated. Also, due to the monolithic nature, ohmic losses are significantly minimized due to the low dispersed current density paths and tunneling between junctions.<sup>40</sup> By using integrated tandem monolithic PEs, the PEC reactor design can be more easily narrowed, which is essential for reducing manufacturing costs and facilitating device installation. Still, monolithic wireless configurations have been less investigated than other electrode configurations, which is mainly due to the

complexities and difficulties related to electrode fabrication and characterization. The characterization of a wireless monolithic device is particularly challenging, as it does not allow having any direct electrical wiring to a potentiostat. Therefore, the performance of the device can only be assessed by collecting the gaseous products, which is experimentally more difficult, as gas leakage, especially H<sub>2</sub>, is one of the most reported issues in lab-scale devices.<sup>229</sup> Notwithstanding, monolithically integrated PEs offer unique advantages for achieving commercial PEC water splitting, due to the simplicity and compactness of the associated device. Ahmet *et al.* have demonstrated the use of low-cost and scalable deposition techniques to produce a planar and nanostructured tandem photoanode, consisting of n-Si/SiO<sub>x</sub>/TiO<sub>2</sub>/WO<sub>3</sub>/BiVO<sub>4</sub>/Fe(Ni)OOH and a Pt cathode as an H<sub>2</sub> evolution catalyst, which provided a combined photopotential capable of unassisted solar water splitting, generating 0.3 mA cm<sup>-2</sup> in 1.0 M KBi pH 9.3 buffer solution.<sup>230</sup> Zhu *et al.* have also demonstrated a monolithic PEC device with simple engineering and low loading of catalysts through the integration of a new class of double perovskite cobaltites for the OER, reporting a 6.6% STH under 71 h of



**Table 4** General overview of PEC devices built and tested over time: detailed information concerning large-scale design features; ✓ – successfully implemented/optimized; ◐ – not assessed; ◐ – non-optimized or not possible to implement

Year	Author designation	Design type (according to Fig. 7)	Photoactive area (cm <sup>2</sup> )	Design features	General indicators	Ref.
2011	Stacked PEC reactor	Type I	<ul style="list-style-type: none"> <li>• Demonstrated: – (10 × 10) cm<sup>2</sup></li> </ul>	<ul style="list-style-type: none"> <li>✓ Gas separation: Nafion membrane</li> <li>✓ PV integration: a PV cell can be placed outside the PEC reactor or immersed in the electrolyte using protective materials</li> <li>◐ Electrolyte feeding: not optimized for continuous operation</li> <li>◐ Configuration of electrodes: not suitable for tandem arrangements (photoanode//HER metal catalyst)</li> <li>◐ Concentrated sunlight: not assessed.</li> </ul>	<ul style="list-style-type: none"> <li>• PE: Si-doped α-Fe<sub>2</sub>O<sub>3</sub></li> <li>• Performance: 2.5 mA cm<sup>-2</sup>@ 0.7 V<sub>HgO Hg</sub> in 1 M NaOH under white (Xe) lamp irradiation (35 W m<sup>-2</sup>)</li> <li>• Stability: not assessed.</li> </ul>	215,216
2014–2016	PortoCell	Type I	<ul style="list-style-type: none"> <li>• Demonstrated: – 2014 version: (10 × 10) cm<sup>2</sup></li> <li>– 2016 version: (7.2 × 7.2) cm<sup>2</sup></li> <li>• Prototype: – 4 PEC cell modules of (7 × 7) cm<sup>2</sup></li> </ul>	<ul style="list-style-type: none"> <li>◐ Gas separation: Teflon<sup>®</sup> or ion exchange membranes; gas crossover was observed when the device was operated in a tilted position</li> <li>✓ PV integration: PV cell placed behind the PE</li> <li>✓ Electrolyte feeding: fully optimized in the 2nd device version (2016)</li> <li>◐ Configuration of electrodes: not ideal for tandem arrangements (photoanode//photocathode – dual side illumination using a dual mirror system; photoanode//HER metal catalyst; and photocathode//OER metal catalyst)</li> <li>◐ Concentrated sunlight: assessed in 2016 (2nd device version – the front window slightly overheated; further adjustments are necessary).</li> </ul>	<ul style="list-style-type: none"> <li>• PE: α-Fe<sub>2</sub>O<sub>3</sub></li> <li>• Performance: 0.5 mA cm<sup>-2</sup>@1.45 V<sub>RHE</sub> in 1 M NaOH under AM 1.5G illumination (1000 W m<sup>-2</sup>); STH efficiency not presented for tandem configuration</li> <li>• Stability: not assessed.</li> </ul>	39,217,220
2014–2016	EMPA PEC reactor – BioPEC	Type III	<ul style="list-style-type: none"> <li>• Demonstrated: – (10 × 10) cm<sup>2</sup></li> </ul>	<ul style="list-style-type: none"> <li>◐ Gas separation mechanism was not specified</li> <li>✓ PV connection: a PV cell can be placed outside the PEC reactor or immersed in the electrolyte using protective materials or tandem PEs</li> <li>◐ Electrolyte feeding system was not investigated</li> <li>✓ Configuration of electrodes suitable for tandem arrangements (photoanode ⊥ HER metal catalyst and photocathode ⊥ OER metal catalyst)</li> <li>◐ Concentration sunlight: not assessed.</li> </ul>	<ul style="list-style-type: none"> <li>• PE: α-Fe<sub>2</sub>O<sub>3</sub> photoanode</li> <li>• Performance 0.45 mA cm<sup>-2</sup>@1.45 V<sub>RHE</sub> in 1 M KOH under AM 1.5G illumination (1000 W m<sup>-2</sup>)</li> <li>• Stability 70 min running time</li> </ul>	221–223
2015	ArtipHyction prototype	Type I	<ul style="list-style-type: none"> <li>• Demonstrated: – 1.6 m<sup>2</sup>, composed of 20 strings of 5 PEC cell modules of (58.0 × 12.4) cm<sup>2</sup></li> </ul>	<ul style="list-style-type: none"> <li>✓ Gas separation: Nafion membrane</li> <li>✓ PV integration: PV cells were placed outside the PEC reactor (the overall exposed illuminated area was increased; the final configuration is not a real tandem system)</li> </ul>	<ul style="list-style-type: none"> <li>• PE: BiVO<sub>4</sub> photoanode</li> <li>• Performance: 2% STH efficiency</li> </ul>	218





Table 4 (continued)

Year	Author designation	Design type (according to Fig. 7)	Photoactive area (cm <sup>2</sup> )	Design features	General indicators	Ref.
2015	Louvered cell	Type II	<ul style="list-style-type: none"> <li>• Demonstrated:               <ul style="list-style-type: none"> <li>– 2 PEC cell modules of (1.6 × 4.4) cm<sup>2</sup> – possible to increase</li> </ul> </li> </ul>	<ul style="list-style-type: none"> <li>✓ Electrolyte feeding: assessed</li> <li>• Configuration of electrodes: not suitable for tandem arrangements (photoanode//HER metal catalyst and photocathode//OER metal catalyst)</li> <li>• Concentrated sunlight: not assessed.</li> <li>✓ Gas separation: Nafion membrane (only works in acidic media due to electrode constraints)</li> <li>✓ PV integration: PV immersed in the electrolyte (tandem PE)</li> <li>✓ Electrolyte feeding: assessed through multi-physics modelling</li> <li>• Configuration of electrodes: not ideal for tandem arrangements (photoanode//photocathode)</li> <li>• Concentrated sunlight: not assessed</li> <li>• Dual side illumination: no information concerning the system transparency.</li> <li>✓ Gas separation: H<sub>2</sub> and O<sub>2</sub> in separate cells</li> <li>✓ PV integration: the PV cell can be placed outside the PEC reactor, or immersed in the electrolyte using protective coatings or tandem PEs</li> <li>• Electrolyte feeding: not assessed</li> <li>✓ Configuration of electrodes: suitable for tandem arrangements</li> <li>• Concentrated sunlight: not assessed, but it was tested under natural sunlight in outdoor conditions.</li> </ul>	<ul style="list-style-type: none"> <li>• Stability: Not clearly described.</li> <li>• Performance: 0.24% STH efficiency</li> <li>• Stability: 20 h.</li> <li>• Electrodes: Ni foil and Pt-plated stainless-steel mesh</li> <li>• Performance (STH efficiency):               <ul style="list-style-type: none"> <li>1 cm<sup>2</sup>: 7.5%</li> <li>100 cm<sup>2</sup>: 0.68%</li> </ul> </li> <li>• Stability:               <ul style="list-style-type: none"> <li>1 cm<sup>2</sup>: 1 h; 100 cm<sup>2</sup>: 83 h.</li> </ul> </li> <li>• PE: α-Fe<sub>2</sub>O<sub>3</sub> and dual CoPi/W:BiVO<sub>4</sub></li> <li>• Performance (STH efficiency):               <ul style="list-style-type: none"> <li>α-Fe<sub>2</sub>O<sub>3</sub>: 1%</li> <li>W:BiVO<sub>4</sub>: 2%</li> </ul> </li> <li>• Stability:               <ul style="list-style-type: none"> <li>α-Fe<sub>2</sub>O<sub>3</sub>: 1008 h.</li> </ul> </li> <li>• PE: 32 small α-Fe<sub>2</sub>O<sub>3</sub> PEs or single 50 cm<sup>2</sup> planar α-Fe<sub>2</sub>O<sub>3</sub> PEs, connected in parallel</li> <li>• Performance</li> </ul>	224–226
2016–2020	Membrane-free PEC cell for alkaline water electrolysis in separate H <sub>2</sub> and O <sub>2</sub> cells	Type IV	<ul style="list-style-type: none"> <li>• Demonstrated:               <ul style="list-style-type: none"> <li>– 1 and 100 cm<sup>2</sup></li> </ul> </li> <li>• Forecasted:               <ul style="list-style-type: none"> <li>– 5 × 9 array of (10 × 10) cm<sup>2</sup></li> </ul> </li> </ul>	<ul style="list-style-type: none"> <li>• Demonstrated:               <ul style="list-style-type: none"> <li>– 1 and 100 cm<sup>2</sup></li> </ul> </li> <li>• Forecasted:               <ul style="list-style-type: none"> <li>– 5 × 9 array of (10 × 10) cm<sup>2</sup></li> </ul> </li> </ul>	<ul style="list-style-type: none"> <li>• Electrodes: Ni foil and Pt-plated stainless-steel mesh</li> <li>• Performance (STH efficiency):               <ul style="list-style-type: none"> <li>1 cm<sup>2</sup>: 7.5%</li> <li>100 cm<sup>2</sup>: 0.68%</li> </ul> </li> <li>• Stability:               <ul style="list-style-type: none"> <li>1 cm<sup>2</sup>: 1 h; 100 cm<sup>2</sup>: 83 h.</li> </ul> </li> <li>• PE: α-Fe<sub>2</sub>O<sub>3</sub> and dual CoPi/W:BiVO<sub>4</sub></li> <li>• Performance (STH efficiency):               <ul style="list-style-type: none"> <li>α-Fe<sub>2</sub>O<sub>3</sub>: 1%</li> <li>W:BiVO<sub>4</sub>: 2%</li> </ul> </li> <li>• Stability:               <ul style="list-style-type: none"> <li>α-Fe<sub>2</sub>O<sub>3</sub>: 1008 h.</li> </ul> </li> <li>• PE: 32 small α-Fe<sub>2</sub>O<sub>3</sub> PEs or single 50 cm<sup>2</sup> planar α-Fe<sub>2</sub>O<sub>3</sub> PEs, connected in parallel</li> <li>• Performance</li> </ul>	69,192,227
2016–2019	CoolPEC cell	Type III	<ul style="list-style-type: none"> <li>• Demonstrated:               <ul style="list-style-type: none"> <li>– 50 cm<sup>2</sup></li> </ul> </li> </ul>	<ul style="list-style-type: none"> <li>✓ Gas separation: ion exchange membrane</li> <li>✓ PV integration: PV cell placed behind the PE, or immersed in the electrolyte using protective materials or tandem PEs</li> <li>✓ Electrolyte feeding: fully optimized</li> <li>✓ Configuration of electrodes: ideal for tandem arrangements (photoanode ⊥ HER catalyst and photocathode ⊥ OER catalyst)</li> <li>✓ Concentrated sunlight: assessed in the SoCRatus setup.</li> </ul>	<ul style="list-style-type: none"> <li>• Stability:               <ul style="list-style-type: none"> <li>1 cm<sup>2</sup>: 1 h; 100 cm<sup>2</sup>: 83 h.</li> </ul> </li> <li>• PE: α-Fe<sub>2</sub>O<sub>3</sub> and dual CoPi/W:BiVO<sub>4</sub></li> <li>• Performance (STH efficiency):               <ul style="list-style-type: none"> <li>α-Fe<sub>2</sub>O<sub>3</sub>: 1%</li> <li>W:BiVO<sub>4</sub>: 2%</li> </ul> </li> <li>• Stability:               <ul style="list-style-type: none"> <li>α-Fe<sub>2</sub>O<sub>3</sub>: 1008 h.</li> </ul> </li> <li>• PE: 32 small α-Fe<sub>2</sub>O<sub>3</sub> PEs or single 50 cm<sup>2</sup> planar α-Fe<sub>2</sub>O<sub>3</sub> PEs, connected in parallel</li> <li>• Performance</li> </ul>	43,79,217
2018–2020	CoolPEC module	Type III	<ul style="list-style-type: none"> <li>• Demonstrated:               <ul style="list-style-type: none"> <li>– 200 cm<sup>2</sup>, 1 × 4 array of 50 cm<sup>2</sup> PEC cells</li> </ul> </li> </ul>	<ul style="list-style-type: none"> <li>✓ Gas separation: ion exchange membrane</li> <li>• PV integration: If multi-electrode windows are used instead of standard 50 cm<sup>2</sup> PEs, the possibility of placing a PV cell at the back of the PE is compromised.</li> </ul>	<ul style="list-style-type: none"> <li>• Stability:               <ul style="list-style-type: none"> <li>α-Fe<sub>2</sub>O<sub>3</sub>: 1008 h.</li> </ul> </li> <li>• PE: 32 small α-Fe<sub>2</sub>O<sub>3</sub> PEs or single 50 cm<sup>2</sup> planar α-Fe<sub>2</sub>O<sub>3</sub> PEs, connected in parallel</li> <li>• Performance</li> </ul>	97



Table 4 (continued)

Year	Author designation	Design type (according to Fig. 7)	Photoactive area (cm <sup>2</sup> )	Design features	General indicators	Ref.
2019	Integrated PEC (IPEC) device	Type I	<ul style="list-style-type: none"> <li>• Demonstrated:</li> <li>– 4 cm<sup>2</sup>, 4 × 1 cm<sup>2</sup> III-V PV cells</li> <li>– Possible to increase</li> </ul>	<ul style="list-style-type: none"> <li>✓ Electrolyte feeding: fully optimized</li> <li>✓ Configuration of electrodes: ideal for tandem arrangements (photoanode ⊥ HER catalyst and photocathode ⊥ OER catalyst);</li> <li>✓ Concentrated sunlight: assessed in the SoCRatus setup.</li> <li>✓ Gas separation: MEA (proton-exchange nanoporous solid conductor, Nafion 115 membrane coated with catalyst particles on each side, sandwiched between two gas diffusion layers and two bipolar flow plates)</li> <li>✓ Electrolyte feeding: fully optimized</li> <li>○ Configuration of electrodes: may be suitable for tandem arrangements with a buried junction (electronic integration between the photoabsorber and electrocatalyst)</li> <li>✓ Concentrated sunlight: assessed with thermal and mass transport optimization.</li> </ul>	<ul style="list-style-type: none"> <li>2 mA cm<sup>-2</sup>@1.45 V, 1 M KOH, <i>ca.</i> 6000 W m<sup>-2</sup></li> <li>• Stability</li> <li>4 days.</li> <li>• PE: Concept validated with triple junction InGaP/InGaAs/Ge PV solar cell</li> <li>• Performance: 0.88 A cm<sup>-2</sup>, 17.2% STH efficiency, under <i>ca.</i> 474 kW m<sup>-2</sup></li> <li>• Stability: 2 h.</li> </ul>	17

continuous illumination.<sup>231</sup> Monolithic architectures can also be considered for the full integration of conventional PV electrodes in PEC devices, enabling to achieve remarkable STH efficiencies.<sup>21,37,57,232</sup> Kistler *et al.* demonstrated a wireless monolithic vapor-fed PEC device assembled with a III-V triple-junction photovoltaic (PV) electrode embedded in a Nafion membrane, where the catalyst was applied by compression of catalyst-modified carbon electrodes against the front and back PV contacts.<sup>233</sup> This device allowed for achieving an unprecedented 14% STH efficiency over 1000 h of cumulative device operation.

In a side-by-side design (type III), as the nomenclature suggests, CEs are placed side-by-side to the photoactive WE, perpendicular to the direction of the incident light; a membrane is placed between electrodes on both sides to avoid gas mixture – *cf.* Fig. 7. Therefore, the side-by-side architecture is considered more feasible for large-area tandem PEC–PV devices, since it comprises an open pathway for sunlight to reach the sublayers of photoactive electrodes. An example of a demonstrated type III device is the “CoolPEC” cell.<sup>43</sup> This 50 cm<sup>2</sup> device was optimized for continuous operation, including some key features: (i) the PE is simultaneously one window of the cell; (ii) uses a feeding manifold for the electrolyte, assuring efficient gas bubble detachment from the windows, low concentration polarization of the electrolyte and efficient heat dissipation (crucial when the cell is operated under concentrated sunlight); and (iii) a narrow, easy-assemble and cost-efficient embodiment. The “CoolPEC” was successfully demonstrated as a tandem PEC–PV device; when equipped

with a 50 cm<sup>2</sup> α-Fe<sub>2</sub>O<sub>3</sub> photoanode displayed a constant current density of *ca.* 0.45 mA cm<sup>-2</sup> at 1.6 V, under 1000 W m<sup>-2</sup> and with constant electrolyte feeding at 45 °C, over 42 days (1008 h) of continuous operation; the extra bias potential was provided through the combination of two silicon heterojunction (SHJ) solar cells.<sup>43</sup> This cell was also tested with a 50 cm<sup>2</sup> dual CoPi/W:BiVO<sub>4</sub> photoanode assembled in a tandem arrangement with 2-series connected SHJ cells, generating *ca.* 1.7 mA cm<sup>-2</sup>.<sup>79</sup> In 2020, Vilanova *et al.* demonstrated a 200 cm<sup>2</sup> modular device, comprising four 50 cm<sup>2</sup> CoolPEC cells in a linear array; this device was tested outdoors under homogeneously concentrated solar radiation (up to 12.8 kW m<sup>-2</sup>) provided by a linear Fresnel reflector and achieved a maximum current density of *ca.* 4 mA cm<sup>-2</sup>.<sup>97,234</sup> Although the reported STH efficiency for this architecture remains below 10%, this target can be achieved with more efficient PEs, such as GaInP/GaInAs tandem absorbers, which can achieve an STH efficiency of 16%.<sup>235</sup>

Type IV PEC devices consist of a membrane-free water splitting system comprising separated cells for the hydrogen and oxygen evolution, and auxiliary electrodes to ensure the charged species transport between the anode and cathode sites. This architecture, which uses a photoelectrode under alkaline conditions, was disclosed by Technion in 2015<sup>227</sup> and reported by Landman *et al.* in 2017<sup>192</sup> and in 2020.<sup>69</sup> In the demonstrated device, Ni(OH)<sub>2</sub>/NiOOH electrodes, commonly used in rechargeable alkaline batteries, were assembled as auxiliary electrodes (AEs). During electrolysis, the NiOOH electrode is charged in the oxygen cell, while the Ni(OH)<sub>2</sub> electrode is discharged in the H<sub>2</sub>



cell. The oxygen produced at the photoanode can be released to the atmosphere, eliminating the need for sealing the photopanel, while H<sub>2</sub> is produced under unilluminated conditions, in sealed reactors, enabling a centralized production, and avoiding large piping for H<sub>2</sub> collection. This concept was successfully demonstrated in small-area cells (6.03 cm<sup>2</sup>), achieving an STH efficiency of 7.5%.<sup>192</sup> Subsequently, it was upscaled to 100 cm<sup>2</sup> and comprising dual  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanodes assembled in tandem arrangement with a Si-PV module, generating *ca.* 0.55 mA cm<sup>-2</sup> over 10 cycles of 8.3 h. Type IV devices incorporate important cost reduction strategies since the use of membranes is avoided and piping is shortened; even so, the STH efficiency of large-area devices remains too low, and the need for regularly replacing the AEs is a serious drawback. Besides the device architectures considered in Fig. 7 and Table 4, other PEC designs can also be found in the literature.<sup>41,194,236</sup> However, these designs are still at a conceptual stage, due to the lack of experimental validation, or because their scalability is not possible. For instance, a particle-suspension PEC system appears to be a simple and attractive design, with no need for large electrodes, wires and compartmentalization; however, no efficient gas separation method is available for this approach and the price of good-performing particles is prohibitively high.<sup>195,237</sup> Some other reported architectures might prove promising in the near future, but the lack of experimental validation hinders a fair comparison with designs that were already experimentally assessed. Holmes-Gentle *et al.*<sup>41</sup> demonstrated an innovative

approach of a membrane-less PEC cell using multi-physics modelling, where the evolved gases are kept separated by hydrodynamic control; this conceptual design was engineered to minimize energy losses associated with the ionic transport and to avoid membrane costs. PEC designs that consider operation with water vapour have also been proposed, but are still in early stages of development.<sup>194,238</sup> Several patents disclosing PEC reactors have also been filled by different groups/companies over the years; however, in most cases, only the conceptual design is reported, and few report large-area devices.<sup>239–241</sup> Independently from the device architecture, a decision-tree approach is a very effective strategy for designing and optimizing a PEC device, as exemplified in Fig. 8.

### 3.4. Assembling requirements for practical PEC devices

When the commercialization of a technology comes into sight, and particularly in the renewable energy market, discussing an economy of scale becomes inevitable. In this context, parameters such as the modularity and scalability of the devices, their CAPEX, EPC (engineering, procurement, and construction) efforts and, naturally, the cost per unit of mass of produced H<sub>2</sub> turn out to be the differentiating factors when choosing between technology A or B. Although the performance of the device is directly related to the performance of the photoelectrodes, which translates into greater or lesser amounts of H<sub>2</sub> produced by the device, the reactor embodiment assumes an extremely

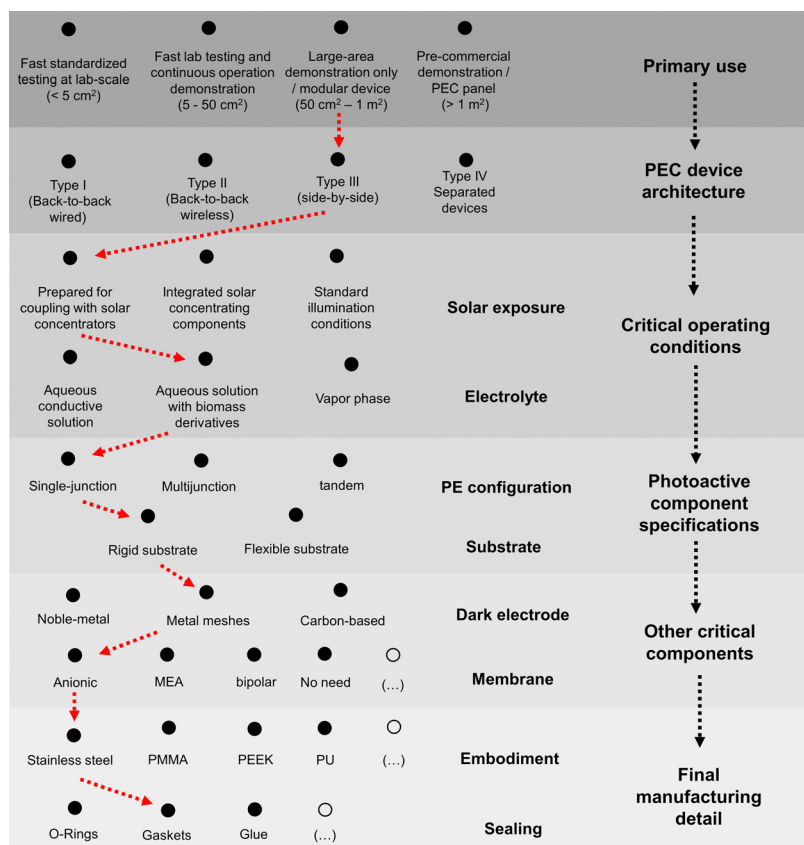


Fig. 8 Example of a decision-tree used to develop a hypothetical and generic new PEC device.





relevant role in the final cost of the technology. On a laboratory scale, the main objective is to build a reactor that is stable to a wide range of electrolytes and that is easily assembled and disassembled. Given these assumptions, most of the reported laboratory devices tend to incorporate expensive materials, such as PEEK,<sup>242</sup> PMMA,<sup>233</sup> and quartz.<sup>97</sup> In an economy of scale, these materials will naturally have to be replaced by cheaper ones commonly used in commercial reactors, such as polyurethane or composite materials. Proper sealing of PEC reactors is also an extremely important aspect for commercialization, both to avoid leakage of corrosive or environmentally harmful liquids and to avoid gas leaks, especially H<sub>2</sub>. Considering PEC devices described in the literature, different sealing strategies have been used, including conventional rubber O-rings,<sup>39</sup> silicone gaskets,<sup>233</sup> and glues.<sup>207</sup> Yet, commercial sealing solutions will most likely be similar to those used in fuel cells and electrolyzers, such as elastomeric gaskets and stack plates. As so, research studies on new PEC devices should take these requirements into consideration and should aim at developing PEC reactors made of low-cost and stable materials already used on a commercial scale, incorporating practical sealing solutions that avoid liquid and gas leakage and do not compromise an easy assembly and replacement of the electrodes. Finally, bearing in mind, the growing environmental restrictions and circular economy directives, the embodiment of commercial PEC reactors should incorporate recyclable, reusable and/or biodegradable materials, such as composite materials or polyurethane.<sup>243–245</sup> Incorporating these materials will be essential to lowering the technology's carbon footprint and ensuring fair green marketing.

## 4. Use of concentrated sunlight and thermal management in PEC devices

The industrialization of PEC-H<sub>2</sub> devices does not rely only on yielding high efficiencies, but also on ensuring a stable and safe operation and reducing production costs. Concentrating solar radiation is a clever strategy for meeting these targets; still, it brings additional challenges, namely related to the increased temperature load and the continuous operation with high current density values.<sup>17,156</sup> Integrating solar concentrators in PEC systems requires further optimization following a symbiotic approach, *i.e.* PEC devices need to be optimized to be operated under concentrated sunlight and solar concentrators need to be redesigned to be compatible with several material restrictions of PEC devices. Also, the operation under an optimum concentration factor depends on the available solar irradiance (*i.e.* varies with the site location) and requires appropriate thermal management solutions for each PEC device architecture.

### 4.1. Overview of concentrating solar technologies

At present, the use of solar concentrators in PEC systems remains almost unaddressed, mainly because large-scale PEC-H<sub>2</sub> production is still at an early stage of development and optimization, with few demonstrated devices. The closest approach to this synergetic strategy at a high TRL is the use of

solar concentrators in PV-electrolyzers.<sup>246</sup> Both large-scale alkaline electrolyzers and PV cells have been optimized, demonstrated and used for industrial H<sub>2</sub> production for many decades and, in more recent years, solar concentrators have been used to further enhance the power output of the PV component.<sup>247,248</sup> Currently, there is no standardized nomenclature in the literature for grouping solar concentrators.<sup>249</sup> Nevertheless, some authors grouped solar concentrators into different categories, usually depending on the operational characteristic that better distinguishes the purpose for which they are used. One option for categorizing solar concentrators considers the type of focal area, which can be linear or punctual.<sup>250</sup> This criterion is directly related to the concentration factor and temperature achieved in the receiver. Linear concentrators provide a homogeneous light concentration along a planar surface or a line, allowing moderate to high-temperature values in the receiver to be reached, usually, ranging from 100 °C to 500 °C.<sup>249,251</sup> On the other hand, punctual concentrators focus sunlight in a single point, allowing very high temperature values in the receiver to be achieved, usually above 500 °C. Another option for categorizing solar concentrators considers the way the solar radiation is focused on the receiver, being reflective or refractive.<sup>252</sup> Caron *et al.* suggested another categorization for solar concentrators suitable for planar PEC devices, based on the type of motion (fixed, translational and rotational) and on the type of tracking (stationary, one-axis and two-axis tracking).<sup>253</sup> Fig. 9 represents schematically the main types of solar concentrators. PEC devices and solar concentrators can be coupled *via* two main assembly arrangements: (i) externally, in separate structures, focusing sunlight only in the PV or PV/PEC components; and (ii) integrated in the device itself, *i.e.* comprised in the glass windows.<sup>254,255</sup> Table 5 qualifies the compatibility of each solar concentrator depicted in Fig. 9 with the PV and PEC technologies, based on multiple categorization criteria.

Reflective concentrators with linear focus are the most common type of solar concentrators, including parabolic trough collectors (PTC), linear Fresnel reflectors (LFR), and compound parabolic concentrators (CPCs). Being studied since the early 1910s, parabolic trough collectors (PTC, Fig. 9(a)) are the most used linear solar concentrators, mainly due to their simple design and low-cost.<sup>251</sup> These concentrators work on a single axis and have a linear receiver. Basically, curved mirrors reflect sunlight towards a dark-coated tube (absorber), which is located just above the mirror and contains the recirculating working fluid.<sup>256</sup> Linear Fresnel reflectors (Fig. 9(b)) are used since the mid-1960s and, like parabolic troughs, operate with a single-axis mirror.<sup>257</sup> These concentrators consist essentially of different arrays of mirrors that focus the solar radiation in a linear central receiver, providing concentration ratios between 30 and 80. The solar receiver is located on the focal line, defined by the tracking mirrors. These mirrors are positioned at different angles and the tracking system is installed at each line of mirrors to concentrate sunlight on the fixed receiver.<sup>258</sup>

Compound parabolic concentrators (CPCs, Fig. 9(c)) are also a mature technology and have been used commercially for a few decades, namely for PV applications, since they enable capturing



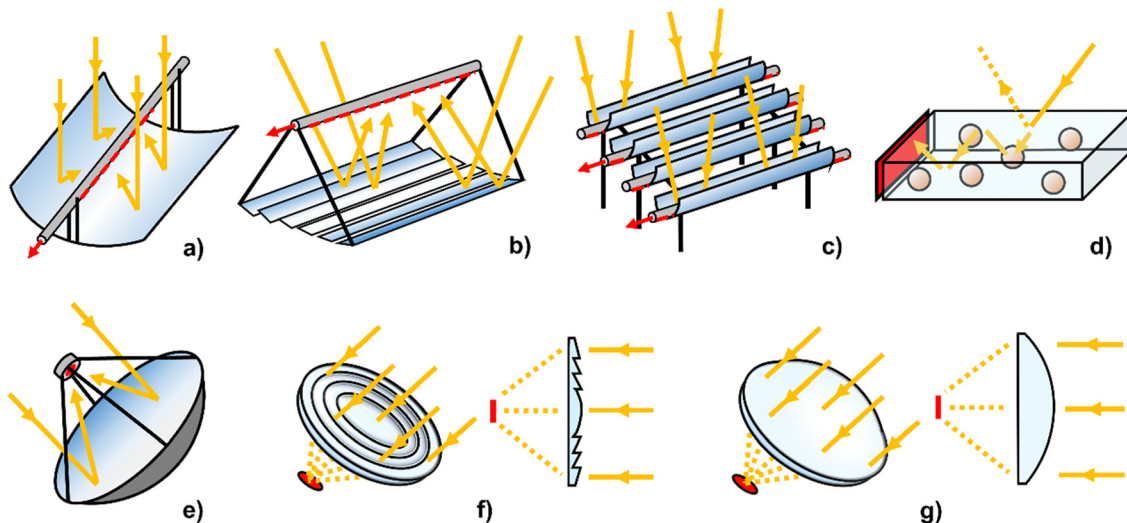


Fig. 9 Schematic representation of the different types of solar concentrators: (a) parabolic trough collector (PTC); (b) linear Fresnel reflector (LFR); (c) compound parabolic concentrator (CPC); (d) luminescent solar concentrator (LSC); (e) parabolic dish concentrator (PDC); (f) Fresnel lens (FL); and (g) convex lens (CL). The focal area in the receiver is highlighted in red.

the diffuse solar radiation together with the direct component. Among CPCs, the most studied and efficient type is the 2-D CPC trough, which allows collecting and concentrating all the rays within a specified acceptance. Some novel features have been added to this type of solar concentrator, namely lens-walled structures, enabling a larger acceptance angle and better optical efficiency. Also, customized 3-D CPCs, ideal for PV applications, are a great improvement over traditional 2-D CPC troughs, achieving optical efficiencies higher than 90%, considering that the incident rays are smaller than the acceptance angle.<sup>259</sup> Advantageous design features of the different reflective linear solar concentrators may be combined to produce better solar concentrators. The best example is the combination of LFRs with CPCs, where the receiver of the LFR incorporates a secondary reflector, usually based on a 2-D CPC.<sup>260</sup> Due to the linear and cylindrical receiver configuration, reflective linear solar concentrators are mainly used for heating a thermal fluid or for steam generation; therefore, they are not usually coupled to water electrolysis devices. Nevertheless, these solar concentrators can be redesigned for coupling with PEC devices, including double-axis tracking systems and special supports for holding linear-array reactors. As a result, these concentrators can provide homogeneous concentrated sunlight in a rectangular focal plane (low to medium concentration range), with the advantage of having a small form factor.<sup>261</sup> SoCRatus, developed by the German Aerospace Centre, is one of the best examples of an LFR suitable for PEC devices, having been already demonstrated for this purpose.<sup>97</sup> This solar concentrator, equipped with 22 linear mirrors and a two-axis tracking system, concentrates sunlight in a rectangular flat focus of  $250 \times 10 \text{ cm}^2$ , ideal for holding linear-array reactors.<sup>43,262</sup>

Solar concentrators with a punctual focus can be considered the second big group of solar concentrating devices, including the reflective and refractive types. Parabolic dish concentrators (PDCs, Fig. 9(e)) are probably the most well-known punctual

solar concentrators, presenting a simple design, based on a parabolic reflective dish, usually equipped with a dual-axis solar tracking system, being commonly used for power generation. The central receiver is located at the focal point,<sup>263</sup> reaching concentration factors up to 3000. PV and PEC devices can be coupled with PDCs, but only if structural changes are made, namely placing the receiver out of the focal point, introducing secondary reflectors or using materials with lower reflectance in the dish.<sup>264,265</sup> Still, by changing the optimal operating parameters of the PDC, the receiver will not get homogenous radiation, which can lead to the appearance of parasitic species transport paths and recombination losses.<sup>266</sup> The concentration of solar radiation in a punctual focus area can also be achieved using Fresnel lenses (Fig. 9(f)) or convex lenses (Fig. 9(g)). This technical solution is mainly used in small devices or modular devices comprising small individual cells. In fact, these types of lenses have been used in concentrated photovoltaic electrolyzers (CPVE) for a few decades.<sup>267</sup> In this case, lenses are typically used to focus sunlight on a special tailored multi-junction solar cell, which is then connected to the anode of the electrolyser.<sup>268</sup> Overall, both types of lenses can be coupled to PEC devices, directing the concentrated solar radiation to the photoelectrode, or integrated in the device itself, as a front window.<sup>269</sup>

Finally, luminescent solar concentrators (LSCs, Fig. 9(d)) are a distinct type of solar concentrating devices, with a different working principle comparatively to the other types. Typical LSCs consist of plastic optical waveguides doped with fluorophores, or glass windows coated with active layers of emissive materials, which can absorb direct and diffused sunlight and then re-emit it at longer wavelengths. Total internal reflection propagates the luminescence towards a receiver (normally a PV cell), placed on the opposite side.<sup>270</sup> LSCs combine the advantage of reflective and refractive solar concentrators – collecting sunlight from a larger area and focusing onto a smaller area; as so, LSCs are ideal for windows.<sup>271</sup> These devices have been pointed





Table 5 Main characteristics of the solar concentrators schematized in Fig. 9 and the assessment of compatibility with PV and PEC devices

Categorization	Solar concentrators (abbreviation)							CL (g)
	PTC (a)	LFR (b)	CPC (c)	LSC (d)	PDC (e)	FL (f)		
Working principle	●	●	●	●	●	○	○	○
Focal area	○	○	○	●	○	○	○	●
Receiver	○	○	○	○	○	○	○	○
Compatibility	○	○	○	○	○	○	○	○
Association with PV/PEC	○	○	○	○	○	○	○	○
Main application	○	○	○	○	○	○	○	○
Defining operating parameter	○	○	○	○	○	○	○	○
Fluid temperature (°C)	350–400; up to 650	100–300; up to 600	80–250	NA*	NR** up to 3000	NR** 60–260; up to 720	NR** 500; up to 900	
Concentration factor (x)	NR**	30–80	NR**	up to 100				

\*NA – non-applicable. \*\*NR – non-relevant.

as a cost-effective alternative to optic-based solar concentration systems aiming at enhancing the performance of stand-alone solar cells and allowing their integration into existing or newly constructed buildings, *i.e.* semi-transparent PV windows capable of transforming energy-passive façades into large-area energy generation units, a particularly appealing strategy towards nearly zero-energy buildings.<sup>272,273</sup> LSCs can be coupled to PEC units, replacing one of the transparent windows. Additionally, if the emission wavelength of the LSC emitters matches the spectral absorption peak of the photo-active electrode, the overall efficiency can be further increased.<sup>274,275</sup>

Concentrated photovoltaic electrolyzers (CPVEs) are the closest available technology to future systems that combine solar concentrators with PEC devices. CPVEs have been investigated as a thriving way for low-cost and clean H<sub>2</sub> production from sunlight, aiming at improving the overall performance of existing solar-powered electrolyzers. Design guidelines have been presented for conceptual designs, based on performance modelling. Based on modelling calculations and LCA analysis, Dumortier *et al.*<sup>276</sup> showed that the use of an integrated concentrated PV electrolyser allows reducing the GHG emissions of H<sub>2</sub> production by up to seven times and increasing the energy yield ratio – the ratio between the energy generated during the lifetime of the equipment and the energy that they incorporate – by four times. A good example of a successfully demonstrated CPVE device was reported by Jia *et al.*, who reported a CPVE device comprising a PEM electrolyser assembled in series with an InGaP/GaAs/GaInNAsSb triple-junction solar cell. This device was operated under concentrated sunlight, at *ca.* 42-suns, where the maximum power point of the PV was adjusted to the operating capacity of the electrolyzers, achieving an STH efficiency of 30% over 48 h.<sup>277</sup> Still, it is important to keep in mind that in CPVEs, solar concentrators and tracking systems will represent a significant part of the operational costs, which will affect the final H<sub>2</sub> production cost. Fresnel and convex lenses are also often used in CPVE applications. Price *et al.*<sup>278</sup> demonstrated that using high-efficiency micro PV cells embedded between a pair of plastic lens arrays allowed reaching a 200-fold increase in the power output comparatively to an unconcentrated cell, at an optical efficiency of 70%, for incidence angles of *ca.* 60°, at a fixed latitude tilt. This work demonstrates that high efficiencies (up to 40%) can be achieved by simply incorporating arrays of convex plastic lenses in the front window and convex mirrors at the back, resulting in a compact and inexpensive design.

Concentrated-PEC (CPEC) systems, coupling two technologies in a single unit, allow for reducing costs and further improving the overall efficiency, since concentrated sunlight is used by both the PV and the electrolyser components.<sup>279</sup> Unfortunately, research work addressing the use of concentrated sunlight in PEC systems is strongly limited by the lack of commercial PEC devices, particularly with an illuminated area larger than 50 cm<sup>2</sup>.<sup>175,199</sup> In a pioneer work by Segev *et al.*,<sup>280</sup> two different types of  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> PES, *i.e.* a 50 nm polycrystalline dense film and a 500 nm mesoporous layer with cauliflower structures, were subjected to simulated concentrated sunlight, ranging from 1 to 25 kW m<sup>-2</sup>. It was observed



that the photocurrent produced by both  $0.28 \text{ cm}^2$  samples increased linearly with solar flux, while the photopotential scaled logarithmically with light intensity. An example of external coupling of a PEC device to a solar concentrator was demonstrated by Khan *et al.*, which coupled a PEC reactor assembled with a triple-junction GaInP/GaInAs/Ge PV cell ( $0.25 \text{ cm}^2$ ), and a Ni foil as an  $\text{O}_2$  evolution catalyst, with a Fresnel lens, kept at a focal distance of 5 cm, achieving an STH efficiency of 13%.<sup>281</sup> Despite the STH efficiency surpassing that assumed as the commercialization target, *i.e.* 10%, the active area of  $0.25 \text{ cm}^2$  remains far from any commercial ambitions. An example of a large-area CPEC device was demonstrated by Vilanova *et al.*<sup>97</sup> The authors reported a  $200 \text{ cm}^2$  PEC modular device designed and optimized for continuous operation under real concentrated sunlight that was coupled with a modified linear Fresnel reflector (LFR). When equipped with nanostructured  $\alpha\text{-Fe}_2\text{O}_3$  PEs, this device generated *ca.*  $2.0 \text{ mA cm}^{-2}$ , below  $14 \text{ kW m}^{-2}$ , corresponding to a hydrogen production of  $5.5 \times 10^{-5} \text{ g}_{\text{H}_2} \text{ h}^{-1} \text{ cm}^{-2}$ , the highest reported for a large-area CPEC device; still, the STH efficiency values varied between 0.36% and 0.12%, which highlights the importance of the compromise between area and efficiency. Splitting the solar spectrum can be a promising strategy to further increase STH efficiency, particularly for combined CPV-PEC devices.<sup>282,283</sup> In the work of Bicer *et al.*,<sup>284</sup> the use of light splitters allowed for achieving power conversion efficiencies of up to 15%, enabling the use of lower wavelengths by a  $\text{Cu}_2\text{O}$  PE.

In general, CPEC devices allow for reaching higher photocurrent densities at lower applied potentials, but they still result from the combination of two separate units, a PEC device and a solar concentrator. Hence, concentrator-integrated PEC (CIPEC) systems can be the ultimate approach since all components are combined in a single unit.<sup>285</sup> The CIPEC approach has been investigated for different PEC systems over the last few years. Zhang *et al.* reported a luminescent solar concentrator based on quantum dots (QDs), which, combined with  $\text{TiO}_2$  photoanodes, achieved a current density of *ca.*  $16.0 \text{ mA cm}^{-2}$  (at  $0.9 \text{ V}_{\text{RHE}}$ ), the highest value reported for PEC cells based on CdS/CdSe QDs; still, the stability was quite low, *i.e.* a 17% photocurrent decrease was recorded after 4 h, attributed to the self-oxidation of QDs induced by ultraviolet radiation.<sup>286</sup> Another example of CIPEC technology was addressed by the SHINE-Nanotera consortium. This project aimed at developing a PEC device with integrated solar concentrator units for operation under medium-range solar concentration, *i.e.* from *ca.* 20 to 50.<sup>287</sup> In this case, the solar concentrator embodied a self-tracking design, with an angular acceptance of  $32^\circ (\pm 16^\circ)$  and an effective concentration of up to 20; water microfluidics was used for thermal management.<sup>288</sup> This smart system used a dichroic mirror to split the solar radiation, where the transmitted part ( $> 750 \text{ nm}$ ) was absorbed by a black paraffin wax layer. Under solar radiation, paraffin melts and expands upwards, creating a coupling feature for the reflected light.<sup>289</sup> Then, the focal spot changes and a different part of the actuator is activated, depending on the position of the sun throughout the day/season. Simpler CIPEC devices can

be assembled by just replacing one window of a PEC reactor with Fresnel or convex lenses. The use of lenses with a short focus distance is compatible with planar PEC device architectures, providing moderate concentration factors, without the need for additional cooling systems.

At an industrial level, the use of concentrated sunlight for PEC- $\text{H}_2$  production is essential to boost  $\text{H}_2$  production and thus easily overcome the claimed disadvantage that PEC systems generate lower current densities than their counterparts. Additionally, the dissipation of excess heat from increased solar load can be more easily achieved in PEC devices than in PVs, since PEC devices comprise electrolyte recirculation by default, which can be optimized for thermal management purposes. Overall, the upscaling of PEC devices for operation under concentrated sunlight must take advantage of the know-how acquired from mature technologies that use concentrated sunlight to drive the water electrolysis, like currently existing CPVE plants.<sup>247</sup>

#### 4.2. Thermal management strategies

Any device that is designed to be operated outdoors, such as PV cells and PEC devices, must incorporate thermal management strategies since, throughout its lifetime, it will experience significant temperature fluctuations, whether related to the ambient temperature, which can vary greatly between seasons of the year, or even throughout the day (depending on the site location), or related to sun exposure. Even under 1-sun, photo(electro)chemical systems can reach working temperatures of *ca.*  $60\text{--}90^\circ\text{C}$ . While excessive heating can reduce the device's lifetime, careful thermal management may improve the overall STH efficiency.<sup>290</sup> Thermal management becomes even more important if coupling with concentrating solar technologies is foreseen, as the receiver area will experience a high thermal load. Taking the example of CPVEs, thermal management was pointed out as the most critical design feature for assuring the expected performance increase due to high photon flux.<sup>291</sup> Also, thermal management can play a determinant role in the final  $\text{H}_2$  cost; it should represent at least 20% of the total power costs, being ascribed mainly to water supply for cooling purposes.

The simplest and most effective strategy to assure proper thermal management in a PEC device is through the optimization of the electrolyte circulation for this purpose. Indeed, water cooling is the most widely used strategy to achieve proper thermal management in multiple applications. By tailor-making electrolyte feeding and recirculation, proper heat dissipation can be achieved, ensuring safe operating conditions that prevent the device from experiencing advanced degradation. As an example, Vilanova *et al.* used fluid mechanics simulation software to design the fluid body of a PEC reactor to optimize the electrolyte flow.<sup>43</sup> Following a stepwise approach by changing the diameter and location of inlets and outlets, the authors developed an electrolyte inlet manifold incorporating several diffusers that allowed reaching an optimized electrolyte upward flow. When subjected to continuous recirculation at  $0.5 \text{ L min}^{-1}$ , the PEC reactor with a photoelectrochemically active area of  $200 \text{ cm}^2$  was demonstrated to effectively maintain a desired operating temperature and dissipate excess heat under concentrated sunlight,



Table 6 Overview of thermal management strategies that can be used in large-scale PEC devices<sup>97,292,293</sup>

Thermal management strategy	Brief description	Pros/benefits	Cons/challenges
Liquid cooling	The simplest and most used strategy. Consists of using the electrolyte flow inherent to a PEC device to cool it. The reactor is designed for having an electrolyte flow that assures proper heat dissipation.	<ul style="list-style-type: none"> <li>– Highly effective</li> <li>– Simple integration</li> <li>– Also promotes an effective collection of evolved gases</li> <li>– Great versatility and adaptability to electrolytes and absorber materials</li> </ul>	<ul style="list-style-type: none"> <li>– Requires a tailor-made manifold design</li> <li>– May require a high electrolyte flow rate</li> <li>– High simulation efforts</li> </ul>
Nanofluid cooling	Use of nanofluids, consisting of liquids containing uniformly distributed nanoparticles.	<ul style="list-style-type: none"> <li>– The required cooling area can be reduced by about 10%</li> <li>– Significant improvement of thermal conductivity and diffusivity</li> <li>– Can be an upgrade to standard liquid cooling</li> </ul>	<ul style="list-style-type: none"> <li>– Nanoparticles increase the viscosity of the liquid</li> <li>– Nanoparticles can cause light trapping and react with the electrolyte</li> <li>– Nanoparticles may represent additional challenges to the recirculation equipment, namely pumps</li> </ul>
Evaporative cooling	Refers to phase-change cooling processes, including boiling and evaporation. Heat of vaporization is the basis of the cooling process.	<ul style="list-style-type: none"> <li>– Decreased coolant flow rate</li> <li>– Simpler and more compact reactor layout</li> <li>– Use of latent heat of phase change</li> </ul>	<ul style="list-style-type: none"> <li>– Requires the use of a vapor-phase electrolyte</li> <li>– Two-phase flow instabilities</li> <li>– Not suitable for all photo absorber materials</li> <li>– May increase degradation</li> <li>– Complex design</li> </ul>
Heat spreaders	Heat is first transferred by conduction and subsequently it is dissipated by forced or free convection. Heat spreaders act as heat sinks. Heat spreaders that have high thermal conductivity are employed near the feeding channel.	<ul style="list-style-type: none"> <li>– In large stacks, it reduces the size of the cooling system.</li> <li>– Low-cost graphite-based materials can be used as heat spreaders.</li> </ul>	<ul style="list-style-type: none"> <li>– The thermal conductivity of the cooling plates has to be very high</li> <li>– Difficult integration in conventional PEC device architectures</li> </ul>
Integrated thermoelectric generators (TEG)	An array of p- and n-type doped semiconductors are assembled electrically in series and thermally in parallel to form a thermoelectric generator. A steady temperature gradient is sustained <i>via</i> passive heating, by conveniently placing the commercial TEG between a hot PEC reactor and a room-temperature medium.	<ul style="list-style-type: none"> <li>– Compact design</li> <li>– Waste heat can be used for generating additional bias potential</li> <li>– Allows integrating either single or tandem light absorbers with the TEG in a single optical light path</li> </ul>	<ul style="list-style-type: none"> <li>– Complex design</li> <li>– The energy levels required for overall water splitting limit the choice of materials to a small selection of wide bandgap semiconductors</li> </ul>
Heat pipes	Similar to a heat exchanger, heat pipes with a recirculating thermal fluid are used to promote heat exchange.	<ul style="list-style-type: none"> <li>– Highly effective</li> <li>– Thermal management can be performed in a separate unit, allowing for more accurate control</li> <li>– Cooling can be optimized separately</li> <li>– Fluids with high thermal conductivity can be used, independently from the electrolyte properties.</li> </ul>	<ul style="list-style-type: none"> <li>– Complex design</li> <li>– As the heat pipes need to pass through the reactor, the pipe materials have to be resistant to the electrolyte</li> <li>– Depending on their location, the inclusion of heat pipes might disturb evolved gas collection and might cause shadowing</li> <li>– Passing pipes through the reactor can add sealing challenges</li> </ul>
Air cooling	An air flow is promoted perpendicular to the PEC reactor to remove excess heat.	<ul style="list-style-type: none"> <li>– Thermal management can be performed in a separate unit, allowing for more accurate control</li> <li>– Cooling can be optimized separately</li> <li>– No special material requirements are needed, namely corrosion resistance</li> </ul>	<ul style="list-style-type: none"> <li>– Added maintenance efforts</li> <li>– Since air has poor thermal conductivity, a high flow rate must be used, increasing power consumption</li> <li>– Colling is promoted from the outside; if the reactor embodiment comprises materials with poor thermal conductivity, such as most polymers, dissipating heat from the inside of the reactor will be difficult</li> <li>– Added maintenance efforts</li> </ul>

at *ca.* 18-suns. Khan *et al.* have also demonstrated that electrolyte recirculation was sufficient to dissipate heat in a tailor-made PEC reactor operating under 42-suns.<sup>281</sup>

Apart from optimized electrolyte recirculation, there are many different thermal management strategies that can be used for PEC applications. Rashidi *et al.* have reviewed some

of the most used thermal management strategies in fuel cells, electrolysers and supercapacitors.<sup>292</sup> From the many different options, those that can be potentially interesting for new PEC devices are summarized in Table 6.

It is important to note that operating under high sunlight concentration requires both thermal and power management



solutions.<sup>287</sup> Since the energy required for promoting the water electrolysis is much higher than the thermodynamic value related to the energy of the reaction, mainly due to high ohmic losses, associated with various components of the cell, *i.e.* the electrodes, the electrolyte, and the membrane, which increase linearly with the generated current but decrease with temperature, thermal management is essential for minimizing performance losses. Tembhurne *et al.* developed a 2-D multi-physics model to investigate and simulate CIPEC devices subjected to highly concentrated sunlight, working at high temperatures and currents.<sup>254</sup> It was concluded that the performance decreased due to the faster degradation of light absorber materials and catalysts in two distinct zones, one corresponding to a low operating current density zone, mostly affected by temperature, and the other corresponding to a high operating current density zone, characterized by mass transport limitations, significantly affected by the maximum power point. The underlying degradation mechanisms can be controlled and counterbalanced with proper thermal and power management, by controlling the electrolyte flow and by installing a power control unit; the authors reported that a 12% degradation in STH efficiency could be compensated by increasing the mass flow rate of the electrolyte by seven times. Tembhurne *et al.* also demonstrated that the recirculation of the aqueous electrolyte solution allows the dissipation of heat from the illuminated semiconductor, preventing photopotential losses.<sup>17</sup>

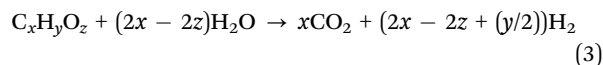
One of the most interesting approaches for efficient thermal management in PEC water splitting under concentrated sunlight was disclosed by Pornrunroj *et al.*<sup>293</sup> The authors reported an integrated thermoelectric (TE)-PEC design for overall water splitting under concentrated solar radiation, where single or tandem light absorbers and heat-harvesting thermoelectric generator (TEG) are assembled in a single optical light path. Moreover, waste heat can be used to generate additional photopotential to accomplish unassisted solar water splitting. A stable temperature can be achieved through passive heating, by placing the TEG between the heated PEC reactor and a room-temperature medium. The fundamental concept relies on assembling electrically in series p- and n-type doped semiconductors, but thermally in parallel to form a thermoelectric generator. The authors were able to report unassisted water splitting under 2-sun by wiring a BiVO<sub>4</sub> PE to a thermoelectric element. Also, the photocurrent of a thermoelectric-perovskite-Fe<sub>2</sub>O<sub>3</sub> system was enhanced 29.7 times under 5-sun. Still, the authors also reported that scalability might be a challenging issue to address for these systems, as in-plane parasitic effects were already noticeable in the reported 6 cm<sup>2</sup> device.

Hence, commercial PEC-H<sub>2</sub> production must consider the use of concentrating solar technologies and thermal management for increasing H<sub>2</sub> production and decreasing performance losses. Therefore, full-scale PEC-H<sub>2</sub> should comprise key elements, as follows: (i) an electrical power station; (ii) a water pre-treatment unit, including filtration systems, reverse osmosis, and a mixing system for adding the electrolyte; (iii) the PEC devices coupled with solar concentrators; (iv) gas-liquid separation systems, one for H<sub>2</sub> and another for O<sub>2</sub> (when O<sub>2</sub> is considered a valuable by-product;

the remaining liquid, should be sent back to the PEC units); and (v) an H<sub>2</sub> purification system, including de-oxidizer and drying sub-systems.<sup>294</sup> Fig. 10 shows a schematic of a hypothetical CPEC-H<sub>2</sub> production plant.

## 5. Use of biomass derivatives in PEC-H<sub>2</sub> production

The use of biomass derivatives has been pointed out as a clever approach to boost H<sub>2</sub> generation in PEC devices, allowing them to overcome thermodynamic and kinetic constraints of conventional PEC water splitting systems and to fulfil circular economy requirements. In these systems, the Gibbs free energy is smaller than that for the water splitting, as the redox potentials of the biomass derivatives are much more negative than that of water, indicating that they are more easily oxidized, making this process more efficient from a thermodynamic point of view.<sup>295</sup> The operating principles of PEC-H<sub>2</sub> production from biomass derivatives and water consist of (i) generation of electron-hole pairs; (ii) movement of electrons from the PE to the CE through an external circuit; (iii) water reduction at the cathode; and lastly (iv) oxidation of biomass by the photogenerated holes to form simpler organic intermediates and CO<sub>2</sub>.<sup>296</sup> The last step (iv) is equivalent to the donation of electrons from the biomass to the PE to consume the photogenerated holes. Overall, the fundamental mechanism of PEC-H<sub>2</sub> production from biomass-derived organics (denoted as C<sub>x</sub>H<sub>y</sub>O<sub>z</sub>) and water can be simplified as follows:



In this case, the performance of the PEC cell is projected to rely on the efficient separation of charge within the PE, on the biomass capability to exchange electrons with the PE, and on ion transport through the electrolyte. Therefore, the addition of organic molecules to the electrolyte started to be explored in the PEC-H<sub>2</sub> field, since it boosts electron transfer due to the smaller activation overpotentials.<sup>297</sup> For PEC devices that use biomass derivatives, H<sub>2</sub> production strongly depends on the properties of the PE, which should fulfil similar key requirements as those of conventional devices: (i) suitable band edge positions; (ii) strong light harvesting ability; (iii) efficient charge transport; (iv) chemical and photoelectrochemical stability; (v) high selectivity and tolerance to poison-species; (vi) low overpotentials for the reduction-oxidation reactions; and (vii) low-cost and abundance. Among the several materials studied for PEC-H<sub>2</sub> using biomass derivatives and water,  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> has been pointed out as one of the most promising materials. This classical semiconductor meets several fundamental requirements, namely a suitable bandgap energy of 1.9–2.2 eV, ideal for solar spectrum harvest, appropriate valence band edge towards water and organic oxidation, and chemical stability in alkaline environments, apart from the well-known abundance, low-toxicity and low-cost. However, it comprises some detrimental features such as low optical absorption due to its indirect bandgap and high recombination rates leading to





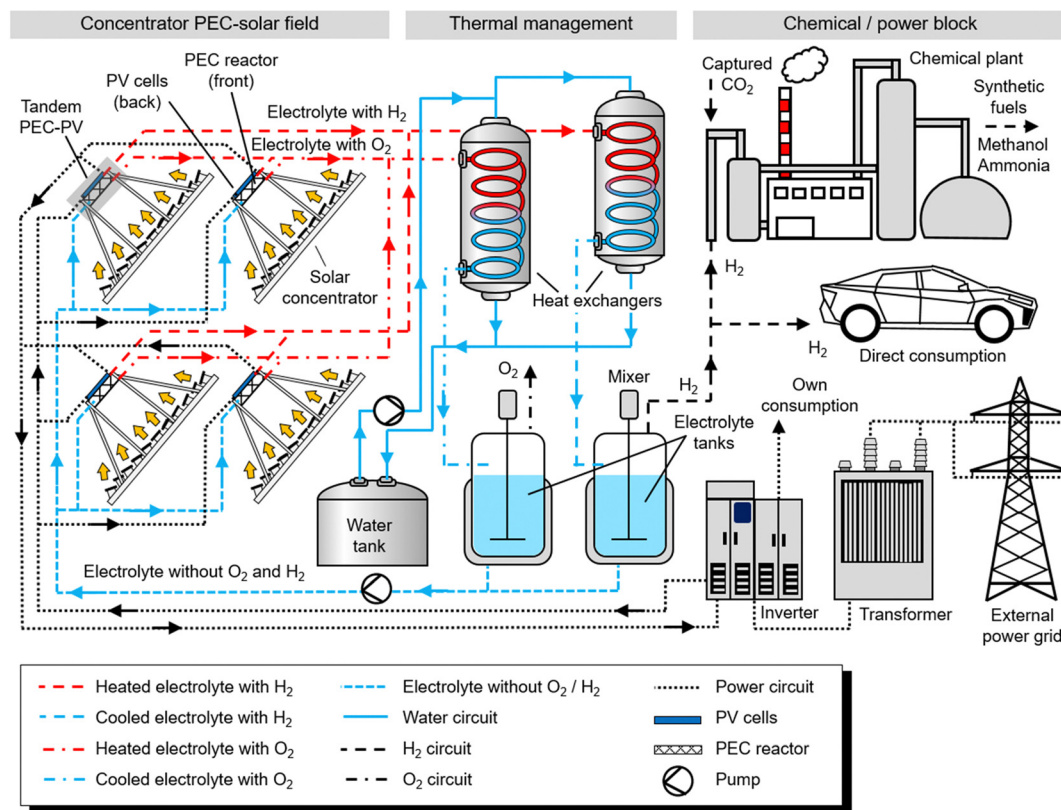


Fig. 10 Schematic representation of a CPEC-H<sub>2</sub> production plant, comprising LFRs coupled with tandem PEC-PV devices and an external thermal management system.

a short hole diffusion length (*ca.* 5 nm). As  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> is the benchmark in stability, but seems to be limited in performance, in the pathway for a highly efficient biomass-driven PEC-H<sub>2</sub> generation device, spinel complex ferrites based on non-CRMs, such as ZnFe<sub>2</sub>O<sub>4</sub> and CuFe<sub>2</sub>O<sub>4</sub>, have also been studied – Fig. 11(a). Both zinc and copper ferrite compounds have favourable bandgap values of *ca.* 2.0 eV and 1.9 eV, respectively, and they can generate higher photocurrent densities under back-side illumination, ideal in a near-zero-gap design. Nonetheless, high recombination rates and low carrier diffusion lengths are critical challenges that have been difficult to overcome. Therefore, upscaled systems should consider clever design features, such as multilayer tailored arrangements, passivation tunnelling, as well as innovative contact architectures, assuring field effect passivation, suitable band alignment, and improved oxidation and reduction processes.

From an environmental point of view, these systems are more competitive than traditional ones, as several biomass by-products, such as methanol and glycerol, are regarded as renewables.<sup>298</sup> Moreover, transforming wastewater and sunlight into green-H<sub>2</sub> breaks the high verticalization of the H<sub>2</sub> market by directly turning an energy consumer into a prosumer. Glycerol is an important by-product of biodiesel and soap production. Particularly in the biodiesel industry, the largest residue is washing wastewater.<sup>299</sup> Biomass-derived biodiesel waste streams can contain glycerol at *ca.* 50% of purity, along

with methanol and organic acids. Also, conventional crops used for glycerol production present a high environmental impact mainly arising from significant land use, especially in the cases of palm oil, soybean oil rapeseed oil and sunflower oil.<sup>297</sup> N-type spinel oxide PES based on MCo<sub>2</sub>O<sub>4</sub> have shown promise, like CuCo<sub>2</sub>O<sub>4</sub>,<sup>300</sup> still, as Co is listed as CRM, CuFeO<sub>x</sub> PE could be more interesting for this application. However, another study on glycerol oxidation on bare catalyst-free  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> photoanodes did not result in the production of formate but originated a variety of products, indicating selectivity challenges.<sup>301</sup> Methanol (MeOH) is another common pollutant in the wastewater from wood pulp, paper, and pharmaceutical industries.<sup>302</sup> Also, the conversion of CO<sub>2</sub> into methanol, a process with particular interest for the upcoming years, usually produces large volumes of wastewater containing trace amounts of MeOH, but enough to prohibit its reuse for electrolysis or as a cooling fluid. Although MeOH is valuable, it is difficult to recover from dilute waste streams, becoming a cheap and valuable resource for PEC-H<sub>2</sub> production. Bare  $\alpha$ -Fe<sub>2</sub>O<sub>3</sub> has been demonstrated to be active for both glycerol and methanol oxidation. Likewise, Zn and Cu ferrites comprise structural motifs that make them promising candidates for biomass oxidation, especially regarding their catalytic activity, as their valence band positions present sufficient thermodynamic driving force.

Nonetheless, using wastewater as an electrolyte in PEC devices is challenging, namely because of the harsh chemical



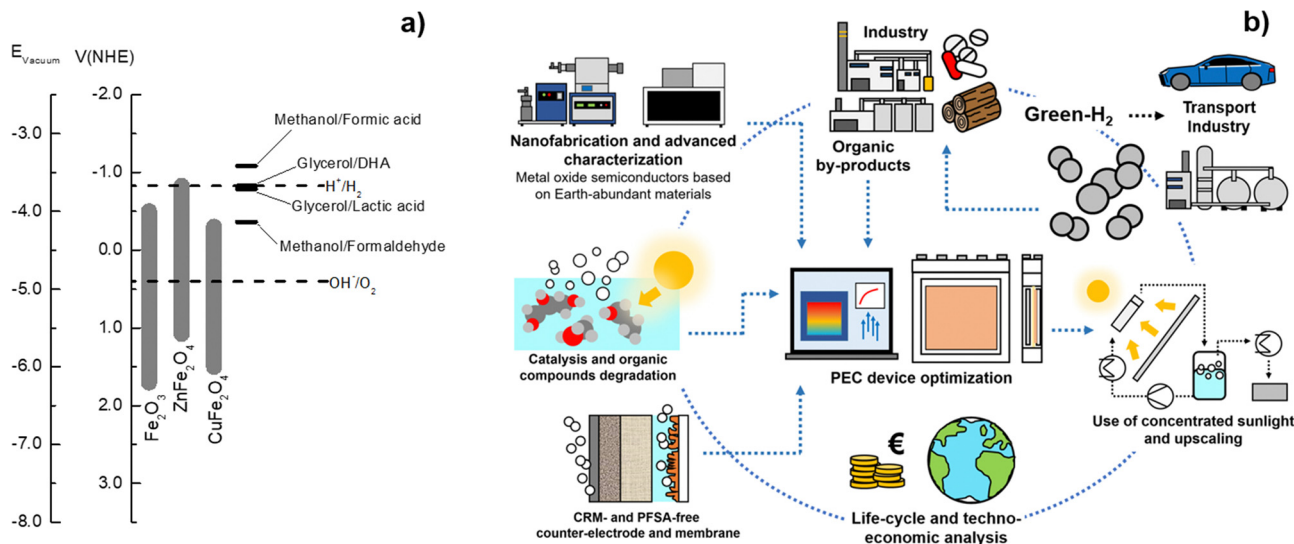


Fig. 11 The use of biomass derivatives for PEC-H<sub>2</sub> production: (a) band alignment of some promising metal oxide semiconductors for PEC-H<sub>2</sub> production intermediated by some industrial organic by-products; and (b) added-value of PEC-H<sub>2</sub> production from biomass derivatives and water to existing supply chains, using industrial by-products, CRM-free electrodes and concentrated sunlight.

environment created (like in the case of MeOH) and added viscosity and clogging (like in the case of glycerol). These challenges can be tackled by (i) using tailored catalyst layers, providing the surface of the PE with “self-cleaning” and anti-sticking capabilities, (ii) optimizing fluid flow, preventing stagnant zones and facilitating the removal of organic deposits, and (iii) using concentrated sunlight, favouring the degradation of organics, both by intensifying photocatalysis and by increasing the thermal load.<sup>303</sup> Fig. 11(b) schematically shows the combination of strategies that should be implemented to achieve high-performing PEC-H<sub>2</sub> production from biomass derivatives and water. As concluded by Huang *et al.*, apart from understanding photocatalytic mechanisms and kinetics at the fundamental level and developing new advanced materials, designing novel photo-reactors able to optimize mass transfer, photon transfer, light utilization, and ready for upscaling is of utmost importance.<sup>304</sup> Moreover, the authors also concluded that optimizing the reactor operating conditions, including photocatalyst loading, light intensity, pressure, pH (liquid phase), reactant feed ratio, *etc.* is equally important for efficient PEC-H<sub>2</sub> production from biomass-derived products. This vision is accompanied by Luo *et al.*, as the authors highlight that the rational design of semiconductor-cocatalyst systems based on Earth-abundant elements is essential for assuring high efficiency and selectivity for a given by-product used as feedstock.<sup>297</sup>

Complementarily, the system as a whole requires a tailored optimization, where reaction parameters should be carefully chosen, namely temperature, feedstock concentration, electrode distance and electrolyte flow. Also, to improve the environmental performance of these systems, the development of a mature market for oxidation products is essential. Another interesting approach has been explored by the research group of Reisner. The authors have been demonstrating the materialization of different circular economy approaches by combining waste

photodegradation with solar fuel production in PEC devices.<sup>305,306</sup> In this case, particulate photocatalysts are often considered, which is not the focus of this review. Nevertheless, Bhattacharjee *et al.* have successfully used perovskite PEs for this purpose.<sup>118</sup> Additionally, the authors have also addressed one of the most important environmental problems of the present, which is the contamination of water resources with plastics;<sup>307</sup> here, plastics can be seen as a raw material for boosting H<sub>2</sub> production in PEC devices. As an example, the authors reported a perovskite-based PEC device capable of simultaneously producing H<sub>2</sub> and reforming soluble biomass and plastic derived, using a novel  $\text{Cu}_{30}\text{Pd}_{70}$  oxidation catalyst, achieving 60–90% product selectivity and sustaining unassisted photocurrents of 4–9 mA cm<sup>-2</sup> for plastic, biomass, and glycerol conversion, in either a two-compartment or integrated “artificial leaf” configuration.

## 6. Overview of techno-economic analysis and life-cycle assessment on H<sub>2</sub> production

Selecting the appropriate H<sub>2</sub>-production technology for industrial applications is a massive challenge. At present, green-H<sub>2</sub> is produced mainly using mature technologies, such as conventional electrolyzers, which are often operated with electricity from the grid. There are few available wide comparative studies on H<sub>2</sub> production methods that combine life cycle and techno-economic indicators, which are essential for understanding how PEC-H<sub>2</sub> production performs from an all-inclusive point of view. Indeed, for PEC-H<sub>2</sub> production, which is at an early stage of development, adopting a holistic approach is vital for improving its competitiveness, regarding both environmental and economic metrics.



Life cycle assessment (LCA) is the most widely used tool for assessing the environmental impacts attributed to a product or a service, comprising three major stages: (i) goal and scope definition, including the selection of life cycle boundaries; (ii) inventory data collection on material and energy flows, where emissions and consumption of resources are identified and quantified; and (iii) estimation of the environmental impacts of each life cycle step, normalization of the impact category indicators and grouping by weighting or significance.<sup>308</sup> The most common type of LCA is the cradle-to-grave approach, which considers all steps of the life cycle, from the extraction of raw materials until the disposal/end-of-life. For energy systems, due to their complexity, a cradle-to-gate approach is more effective, usually including only the production and distribution stages. Particularly for PEC-H<sub>2</sub>, since commercial large-scale PEC devices are not available yet, LCA studies are based mainly on assumptions and extrapolations from existing devices, which were developed for laboratory purposes. This can result in values with no practical significance and are unfairly detrimental to the PEC water splitting technology. Sathre *et al.* performed a close approach to what could be an ideal prospective life-cycle inventory for PEC water splitting devices.<sup>232,309</sup> Based on the detailed inventory, the authors were able to calculate relevant techno-economic indicators: large-scale PEC-H<sub>2</sub> production with a 10% STH efficiency and a life span of 20 years (considering 1 GW per year, equivalent to 610 ton of H<sub>2</sub> per day) would have an energy payback time (EPBT) of 8.1 years, an energy return on energy invested (ERoEI) of 1.7, and a life-cycle primary energy balance of  $5.0 \times 10^{17}$  J, considering 40 years of projected service life. If the STH efficiency is increased to 20%, the EPBT can be further reduced to 3 years.<sup>232</sup> Although Sathre's study focused on net-energy parameters, it was possible to identify the most relevant components of the life cycle inventory: (i) fabrication of active materials; (ii) production of the materials used in the embodiment; (iii) assembly of the cell; (iv) assembly of the module; (v) operation of the module (including auxiliary equipment); and (vi) end-of-life. The STH efficiency and the life span of the PEC cells remain critical parameters, since they affect the net energy balance, whereas the balance of system (BOS) parameters, such as the construction and operation of liquid and gas infrastructures, present a lesser significance. However, the most limiting factor of comparative analyses based on LCA studies is the multiplicity of reported impact categories and the lack of standardization in the way results are presented. Following, the comparative outlook presented here considers only the global warming potential (GWP, kg<sub>CO<sub>2</sub>eq.</sub> kg<sub>H<sub>2</sub></sub><sup>-1</sup>) and acidification potential (AP, g<sub>SO<sub>2</sub>eq.</sub> kg<sub>H<sub>2</sub></sub><sup>-1</sup>) as categories of environmental impacts, since they are the most addressed indicators; the functional unit is 1 kg of generated H<sub>2</sub>.

Similar to environmental performance, assessing the economic viability of H<sub>2</sub> production methods is crucial for having a holistic view of their competitiveness, and especially for understanding how competitive PEC-H<sub>2</sub> is against the other options.<sup>37,38</sup> Due to the lack of standardization of results and ambiguities in the definition of variables and criteria for quantifying economic performance, the payback time and the

levelized cost of hydrogen (LCOH) were the economic indicators selected for the comparative analysis presented here. In short, the payback time indicates the duration for which a facility needs to be operated to produce an amount of H<sub>2</sub> equivalent to the energy required for its manufacturing, construction, and decommissioning; it is calculated as the ratio between the fixed energy input and the annual net energy output under full-scale steady-state operation.<sup>232</sup> This parameter is not suitable for classification as a life-cycle metric, since it does not consider the energy generated after reaching the payback time; therefore, the service life of the facility is not included in the calculations. The LCOH usually refers to the ratio between the net present value of the total cost of constructing and operating an H<sub>2</sub> plant and the total H<sub>2</sub> production over its lifetime. As so, it can be assumed as the total cost needed to bring an H<sub>2</sub> production facility to a commercially operational status, combining individual purchase costs with installation factors, indirect costs (*i.e.* engineering, construction, contingency costs, taxes, *etc.*) and maintenance during the lifetime of the facility.

Finally, it is essential to choose the H<sub>2</sub>-production technologies that should be compared with PEC water splitting. At present, there is no specific taxonomy for grouping H<sub>2</sub> production methods, but one can assume a logical categorization depending on the main energy input, being classified as fossil fuel- or renewable-based.<sup>310,311</sup> Fossil fuel-based technologies are more mature, widely developed and account for almost 95% of current H<sub>2</sub> production.<sup>312</sup> Among them, steam methane reforming (SMR) and underground coal gasification (UCG) are the most widespread; others include hydrocarbon pyrolysis and autothermal reforming. Despite accounting for only 5% of current H<sub>2</sub> production, renewable-based technologies are more diverse but essentially comprise (i) water splitting based technologies, which include electrolyzers (alkaline – AEC, proton exchange membrane – PEM, and solid oxide electrolysis cells – SOEC, supplied with electricity from the grid or from wind, hydro or solar-PV plants), photoelectrochemical (PEC) water splitting and high-temperature electrolysis (HTE), and (ii) bio- and thermochemical processes, where biomass and organic matter are degraded, generating H<sub>2</sub>, which can be assisted or not by solar radiation (direct and indirect bio-photolysis, photo and dark fermentation).<sup>313</sup> A new way of classifying H<sub>2</sub> production methods has emerged in recent years, consisting basically of a colour code.<sup>314</sup> This colour categorization started when H<sub>2</sub> production based exclusively on renewable sources became unanimously recognized as “green hydrogen”. However, given the diversity of H<sub>2</sub> production technologies, there was a need to create a colour code that could also encompass other technologies, distinguishing the working principle, the primary resources, and the generated emissions. Even so, this colour code is not unanimous and, depending on the literature, some colours might have different definitions, thus making the classifications imprecise. For instance, yellow-H<sub>2</sub> can easily be found as referring to PV-electrolysis or grid-fed electrolysis. Fundamentally, there are three colours that are widely used and accepted, and whose definition is not controversial: (i) grey, produced by steam methane reforming, partial oxidation, or autothermal reforming; (ii) blue, produced from fossil fuels,



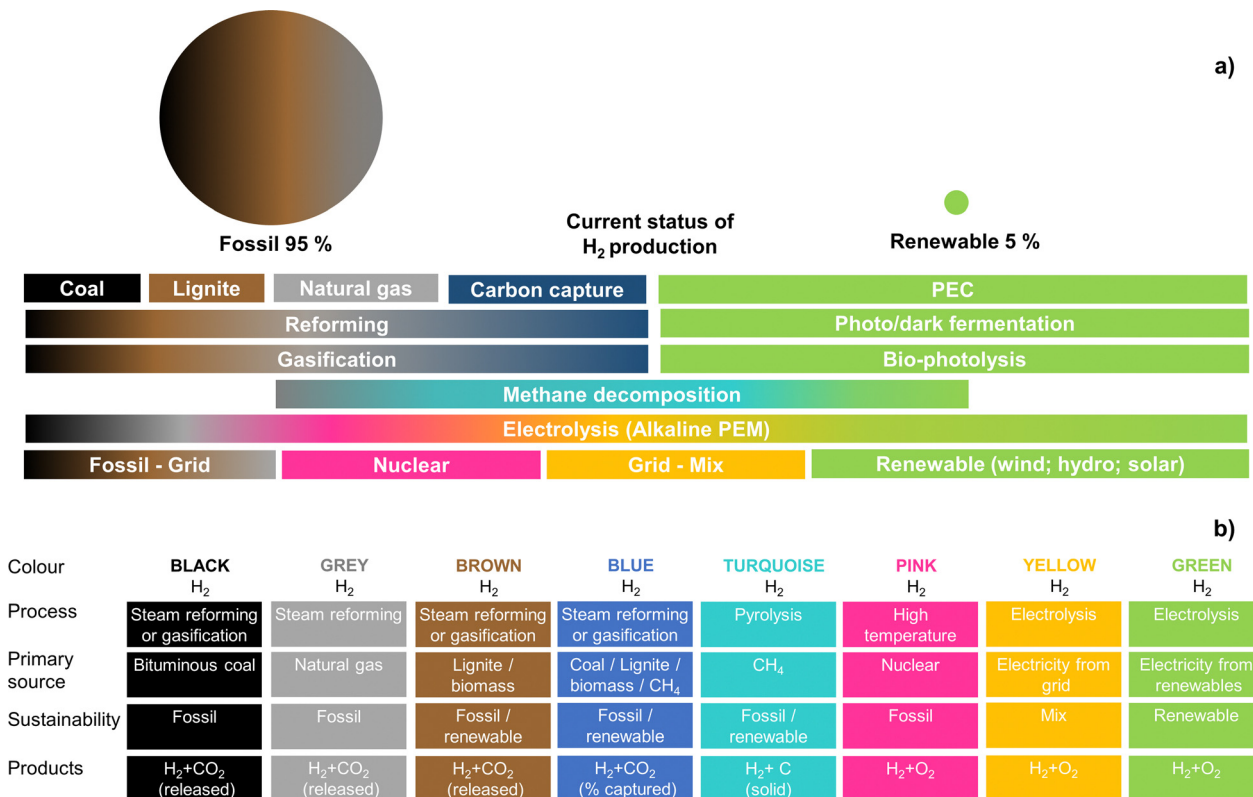


Fig. 12 Grouping and categorization of H<sub>2</sub> production methods: (a) based on renewability; and (b) based on colour code.

but with carbon capture, utilization, and storage; and (iii) green, produced by water electrolysis using exclusively electricity from renewable sources. Fig. 12 schematizes the adopted taxonomy for H<sub>2</sub> production based on renewability and on the H<sub>2</sub> colour code. Although the colour code considered in Fig. 12 might seem something trivial or of little importance, the truth is that it has become extremely relevant in the H<sub>2</sub> market, since this classification has been directly influencing the value of products made from H<sub>2</sub>, such as synthetic fuels, ammonia, methanol, among others. Furthermore, the inconsistency and dynamics of the CO<sub>2</sub> market<sup>315,316</sup> have played a crucial role in the H<sub>2</sub> price, boosting the choice for processes that are based on renewable sources and/or have a neutral or very low carbon footprint.<sup>317</sup> Given the evident trend of increasing taxes and penalties for the emission of greenhouse gases (translated into tons of CO<sub>2</sub> eq.), this has become the present determining criterion for choosing a H<sub>2</sub> production technology. In fact, presently the economic competitiveness of H<sub>2</sub> production processes relies more on the cost of CO<sub>2</sub> emissions rather than on the cost of raw materials.<sup>318,319</sup> Furthermore, this discussion often leads to existentialist questions such as “how green electrolysis really is?” If one considers only the H<sub>2</sub> production process, water electrolysis does not generate CO<sub>2</sub> emissions, but if one considers, for instance, the life cycle of PV panels, the CO<sub>2</sub> balance may not be so green. On the other hand, methane decomposition results in the production of turquoise H<sub>2</sub> which, if based on natural gas, can have a neutral CO<sub>2</sub> balance, but if biomethane is used, the CO<sub>2</sub> balance can be negative. Therefore, it becomes imperative to assess the

environmental performance of a certain H<sub>2</sub> production process based on its life cycle, particularly regarding its global warming potential, as it will strongly influence the LCOH.

For the present comparative overview, a brief description of each H<sub>2</sub> production method, as well as its pros and cons, benefits and challenges, is summarized in Table 7. Only H<sub>2</sub> production technologies for which it was possible to collect enough data to perform a reasonable comparison were considered, as detailed in Table 8. The comparative outlook of environmental impact categories and production costs of the selected H<sub>2</sub> production method is presented in Fig. 13.

The obtained results clearly show that UCG is the H<sub>2</sub> production technology with the worst environmental performance, with a GWP of 18.0 kg<sub>CO<sub>2</sub> eq.</sub> kg<sub>H<sub>2</sub></sub><sup>-1</sup> and an AP of 34.1 g<sub>SO<sub>2</sub> eq.</sub> kg<sub>H<sub>2</sub></sub><sup>-1</sup>. Nevertheless, AEC grid presented the highest GWP, with a total of 24.6 kg<sub>CO<sub>2</sub> eq.</sub> kg<sub>H<sub>2</sub></sub><sup>-1</sup>. It is important to note that this GWP value comes with a significant error, which is due to the highly variable nature of electricity from the grid; in many countries, the grid is supplied by fossil-fuel-powered plants, which significantly hinders the environmental performance of AECs.<sup>342,376</sup> On the other hand, wind-electrolysis presented the lowest GWP contribution, with 1.0 kg<sub>CO<sub>2</sub> eq.</sub> kg<sub>H<sub>2</sub></sub><sup>-1</sup>, while PEC water splitting had the lowest associated AP, with 1.3 g<sub>SO<sub>2</sub> eq.</sub> kg<sub>H<sub>2</sub></sub><sup>-1</sup>. Overall, PEC water splitting is the H<sub>2</sub> production method with the lowest environmental impact, which is consistent with other comparative studies.<sup>390</sup> As an example, Karaca *et al.* performed a cradle-to-gate LCA study considering PEC-H<sub>2</sub> production based on an 870 cm<sup>2</sup> PEC reactor that generated 3.15 mA cm<sup>-2</sup>.<sup>343</sup>





Table 7 Brief description, key benefits and challenges of the H<sub>2</sub> production processes selected for the comparative study

Technology/abbreviation	Brief description	Pros/benefits	Cons/challenges
Underground coal gasification/UCG <sup>320</sup>	Still, one of the most used processes to produce H <sub>2</sub> . Coal is subjected to high temperature and pressure conditions in the presence of a controlled amount of O <sub>2</sub> (or steam), being broken down into syngas, <i>i.e.</i> carbon monoxide (CO), hydrogen (H <sub>2</sub> ), and methane (CH <sub>4</sub> ).	<ul style="list-style-type: none"> <li>– One of the most mature technologies for H<sub>2</sub> production</li> <li>– Low payback time</li> <li>– Low water footprint</li> <li>– Cleaner and more effective than coal combustion (if power generation is considered)</li> <li>– Currently the most common and cost-effective method for H<sub>2</sub> production.</li> </ul>	<ul style="list-style-type: none"> <li>– High emissions of GHG</li> <li>– High acidification potential</li> <li>– As it uses a fossil fuel, the generated H<sub>2</sub> is not renewable</li> <li>– If carbon capture is not considered, the high carbon taxes may undermine cost-effectiveness</li> <li>– High emission of GHG</li> <li>– High capital costs</li> <li>– Coke deposition during the reaction.</li> </ul>
Steam methane reforming/SMR <sup>321,322</sup>	Methane is first catalytically reformed at high temperatures, producing carbon monoxide and a syngas mixture. H <sub>2</sub> is further produced through the water–gas shift reaction.	<ul style="list-style-type: none"> <li>– Abundant, renewable, and cheap feedstock.</li> <li>– It is a renewable process</li> <li>– Promotes forest cleaning, reducing the forest fuel and preventing wildfires</li> <li>– Possibility of also using municipal waste in the process.</li> </ul>	<ul style="list-style-type: none"> <li>– Formation of tar aerosols and ash (leading to slagging, fouling and agglomeration)</li> <li>– As the process generates CO and/or CO<sub>2</sub>, the resulting H<sub>2</sub> is considered brown-H<sub>2</sub>.</li> <li>– Expensive separation and purification units</li> <li>– Not suitable for intermittent operation</li> <li>– Substrate pre-treatment is crucial</li> <li>– It can generate undesirable subproducts</li> <li>– Feedstock quality</li> </ul>
Biomass gasification/BG <sup>323</sup>	One of the main routes for the thermochemical conversion of biomass, occurring at higher temperatures than pyrolysis (700–1200 °C), in the presence of an oxidizing agent, originating mainly gaseous products (syngas, <i>i.e.</i> H <sub>2</sub> and CO).	<ul style="list-style-type: none"> <li>– Biogas is a renewable source of methane</li> <li>– Coupling with commercial digesters, landfills, or wastewater treatment plants.</li> </ul>	<ul style="list-style-type: none"> <li>– Carbon capture and storage</li> <li>– Low tolerance for impurities.</li> <li>– Increased capital cost</li> </ul>
Biogas reforming/BGR <sup>324</sup>	Similar to fossil fuel-based SMR, but biogas is used as a renewable resource for producing H <sub>2</sub> (usually 55–65% CH <sub>4</sub> , and 30–45% CO <sub>2</sub> ).	<ul style="list-style-type: none"> <li>– Despite using fossil fuels, the whole carbon balance can be almost zero</li> <li>– Mature process</li> </ul>	<ul style="list-style-type: none"> <li>– It is still non-renewable</li> <li>– Carbon capture is quite expensive</li> <li>– Still at low TRL; few commercial solutions available</li> <li>– Innovative catalysts and reactors are needed for the reaction to occur at lower temperature</li> </ul>
Steam methane reforming with carbon capture and storage/SMR-CCS	The same technology as conventional SMR, but with a carbon capture and storage unit downstream.	<ul style="list-style-type: none"> <li>– H<sub>2</sub> is easily purified</li> <li>– If bio-methane is used, the carbon balance might be negative</li> </ul>	<ul style="list-style-type: none"> <li>– Few chemically stable materials for high temperature and highly reducing/oxidizing environment.</li> <li>– It carries all the well-known problems of nuclear energy, such as safety issues, hazardous waste handling, and the discussion of whether it is renewable or not.</li> </ul>
Methane splitting/MS <sup>325</sup>	Methane decomposition, also known as methane pyrolysis or methane cracking, is an endothermic process (that occurs typically at 700–1100 °C) where CH <sub>4</sub> is decomposed in the presence of a catalyst, generating H <sub>2</sub> and solid carbon.	<ul style="list-style-type: none"> <li>– Absence of GHG emissions</li> <li>– Coupling to existing nuclear power plants allowing large-scale production</li> <li>– Free of GHG emissions.</li> </ul>	<ul style="list-style-type: none"> <li>– Combined heat and gas generation</li> <li>– Mature and reliable technology</li> <li>– Already optimized for large-scale operation</li> <li>– Required constant power supply, preventing intermittent operation.</li> </ul>
High-temperature electrolysis/HTE <sup>326</sup>	Water splitting occurs at high temperatures (700–1000 °C) and the primary source (nuclear, concentrated solar power) generates both thermal and electrical inputs. Solid oxide electrolysis cells (SOECs) are a commonly used process.	<ul style="list-style-type: none"> <li>– High capital cost</li> <li>– Corrosive operational environment</li> </ul>	<ul style="list-style-type: none"> <li>– Operation at high potential (1.8–2.4 V)</li> <li>– Not necessarily green</li> <li>– Need for ultrapure water (leading to the formation of a concentrated wastewater stream)</li> <li>– Two separate devices are required to drive water splitting (PV unit and electrolyser)</li> <li>– Single-bandgap PV cell</li> <li>– Considerable land footprint</li> </ul>
Alkaline electrolysis <i>via</i> electricity from the grid/AEC-grid <sup>327</sup>	Electric current is provided to two electrodes immersed in a liquid electrolyte, usually concentrated KOH, resulting in the production of H <sub>2</sub> at the cathode and O <sub>2</sub> at the anode. The two electrodes are separated by a selective gas diaphragm.	<ul style="list-style-type: none"> <li>– Comprises two mature technologies already optimized for large-scale operation</li> <li>– The main energy source is renewable and clean.</li> </ul>	
Photovoltaic electrolysis/PV-EC <sup>322,328</sup>	Solar panels generate electricity, feeding an electrolyser (PEM or alkaline) with the appropriate potential to split water.		



Table 7 (continued)

Technology/abbreviation	Brief description	Pros/benefits	Cons/challenges
Wind electrolysis/ Wind-EC <sup>329</sup>	Wind power plants (grid-connected or autonomous) feed the electrolyser (directly or through batteries).	<ul style="list-style-type: none"> <li>– The main energy source is renewable</li> <li>– Comprises two mature technologies already optimized for large-scale operation</li> </ul>	<ul style="list-style-type: none"> <li>– Intermittence of sunlight</li> <li>– Intermittence of wind</li> <li>– High capital cost</li> </ul>
Photoelectrochemical/ PEC <sup>44,97</sup>	The electrodes are in contact with an electrolyte in a single device where, at least, one is photoactive. H <sub>2</sub> and O <sub>2</sub> can be produced in the same compartment (divided by an ionic exchange membrane) or in physically separate compartments.	<ul style="list-style-type: none"> <li>– Possibility of using concentrated sunlight</li> <li>– Low-cost materials</li> <li>– Operation at low temperature</li> <li>– Operation at low potentials</li> <li>– Plenty of room for optimization.</li> </ul>	<ul style="list-style-type: none"> <li>– Aesthetic impact and noise.</li> <li>– Still at an early development stage</li> <li>– Scalability and stability of PEs</li> <li>– Operation in a corrosive environment (strong alkaline or acidic media).</li> </ul>

Table 8 Average values of global warming potential (GWP), acidification potential (AP), energy payback time (EPBT), cost and efficiency for each H<sub>2</sub> production method, obtained from a literature survey on LCA and techno-economic studies

	GWP	AP	EPBT	LCOH	Efficiency
	kg <sub>CO<sub>2</sub> eq.</sub> kg <sub>H<sub>2</sub></sub> <sup>-1</sup>	g <sub>SO<sub>2</sub> eq.</sub> kg <sub>H<sub>2</sub></sub> <sup>-1</sup>	Years	kg <sub>H<sub>2</sub></sub> <sup>-1</sup>	%
UCG	18.0 <sup>330–333</sup>	34.1 <sup>330,331</sup>	6.9 <sup>334</sup>	0.9 <sup>330</sup>	72 <sup>330,335</sup>
SMR	16.6 <sup>14,336–343</sup>	16.6 <sup>14,337,339,341–343</sup>	4.6 <sup>344</sup>	1.5 <sup>329,337,342,345–347</sup>	65.5 <sup>348,349</sup>
BG	4.3 <sup>326,350–352</sup>	21.2 <sup>330,351–354</sup>	4.8 <sup>350</sup>	3.5 <sup>330,355–357</sup>	45.0 <sup>330,350</sup>
BGR	7.6 <sup>337–340</sup>	4.9 <sup>345,358,359</sup>	8.0 <sup>360</sup>	3.6 <sup>337,347,361</sup>	44.3 <sup>347</sup>
SMR + CCS	2.3 <sup>311,330,362</sup>	16.8 <sup>353</sup>	7.5 <sup>363</sup>	1.6 <sup>311,330,362,364</sup>	65.8 <sup>330</sup>
MS	5.4 <sup>365–367</sup>	14.7 <sup>365</sup>	n.a.	2.4 <sup>325,366–368</sup>	58.0 <sup>369</sup>
HTE	1.4 <sup>326,339,342,351,352,370</sup>	3.5 <sup>339,342,351,352,359,370</sup>	10 <sup>371</sup>	4.5 <sup>330,342,372,373</sup>	82.0 <sup>330,374,375</sup>
AEC-grid	24.6 <sup>341,376–379</sup>	1.9 <sup>330,353,376</sup>	3.1 <sup>380</sup>	7.2 <sup>327,342,346,352,357,372</sup>	74.2 <sup>341,352</sup>
PV-EC	3.4 <sup>14,337,342,351,352,381</sup>	9.0 <sup>14,337,342,351,352</sup>	10.0 <sup>382</sup>	7.5 <sup>279,330,337,373</sup>	12.3 <sup>277,383</sup>
Wind-EC	0.9 <sup>14,337,339,341,342,352,381,384</sup>	3.7 <sup>14,337,339,341,342,352</sup>	6.4 <sup>385,386</sup>	5.1 <sup>329,337,342,347,386,387</sup>	31.0 <sup>347,358</sup>
PEC	1.0 <sup>29,330,343,351,352</sup>	2.0 <sup>330,343,351,352</sup>	5.4 <sup>232,309,388</sup>	6.9 <sup>29,330,352,373,389</sup>	16.1 <sup>84,232</sup>

The results obtained for PEC-H<sub>2</sub> production were compared with other three H<sub>2</sub> production processes, namely SMR, PV-EC, and wind-driven PEM water electrolysis. It was concluded that PEC water splitting is the most environmentally friendly technology, namely considering the GWP and AP impact categories. This study also reveals that the PEC reactor and the membrane are the components with greater contribution to the total environmental impact, which confirms the importance of using non-toxic, abundant, and highly recyclable materials in device design.

As solar-based AEC is the currently available industrial technology that is closer to PEC-H<sub>2</sub> technology, a parallel comparison between LCA studies can provide key information on which components and which lifecycle stages have a higher contribution to the environmental impacts of future PEC-H<sub>2</sub> plants. In an LCA study conducted by Palmer *et al.* on PV-electrolysis, the authors found that most of the normalized environmental impact of the complete system came from the PV panels, mainly related to the production of metallic silicon.<sup>391</sup> Hence, PEC devices can be competitive, as they are not so dependent on metallic silicon and can incorporate photo-absorbing electrodes that are made from Earth-abundant materials and do not require such energy-intensive manufacturing processes. In another LCA study on alkaline water electrolysis, Koj *et al.*

concluded that the anode and the cathode have the highest contribution to the acidification potential (AP), while the infrastructure and the electrodes contribute equally to the GWP.<sup>376</sup> This reinforces the need for choosing non-toxic, abundant, and chemically stable materials for PEC water splitting, produced *via* clean and non-complex fabrication techniques. It was also found that operating an AEC-H<sub>2</sub> plant for 5 years has an impact on AP and GWP that is 9 times higher than the construction stage; as so, optimizing the operating parameters, such as maximum solar harvesting and thermal and power management, becomes critical for reducing the environmental impact of future PEC-H<sub>2</sub> production plants. In terms of economic performance, SMR remains the most competitive option, with a payback time of 4.6 years and a production cost of 1.5 \$ \$ kg<sub>H<sub>2</sub></sub><sup>-1</sup>, although AEC-grid has the lowest payback time, averaging only 3.1 years. However, it is important to note that these results are based on studies that do not consider sudden events that can strongly affect the price of fossil fuels and, particularly, the price of natural gas; as an example, in a timeline of only a few months during 2022, the price of natural gas increased by more than 200%, reaching values as high as 315 € MWh<sup>-1</sup>, which can ruin the competitiveness of SMR.<sup>392</sup> In this scenario, and based only on the cost of thermodynamic energy, SMR would produce H<sub>2</sub> at a cost of 21 USD kg<sup>-1</sup>, and renewable-based H<sub>2</sub> production would



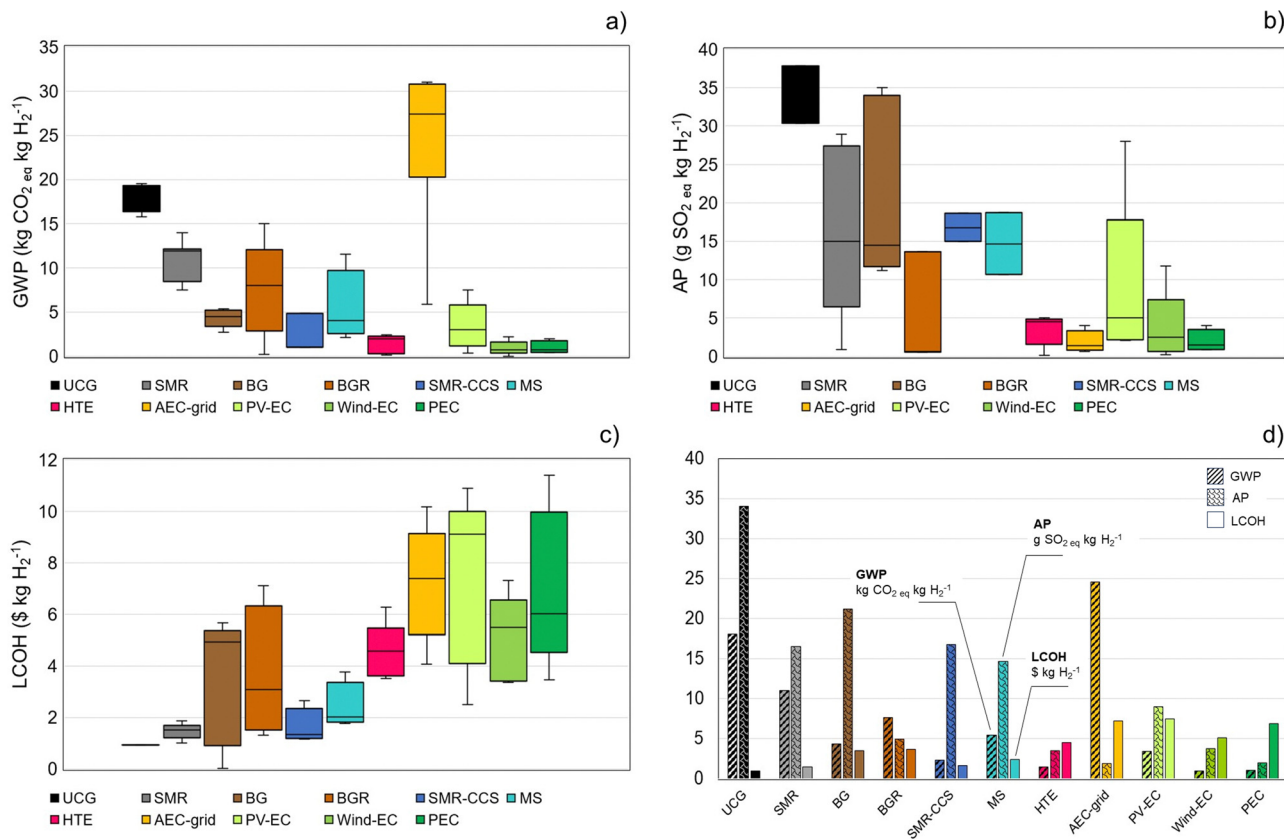


Fig. 13 Comparison of H<sub>2</sub> production processes based on LCA indicators and costs: (a) global warming potential (GWP); (b) acidification potential (AP); (c) levelized cost of hydrogen (LCOH); and (d) global comparison between the previous indicators.

become much more competitive. Nevertheless, and based on a “so-called standard scenario”, AEC-grid achieves the lowest payback time; indeed, it is the first H<sub>2</sub> production technology, studied over two centuries. On the other hand, HTE has the highest payback time, mainly because it is commonly associated with nuclear power plants, which require a large initial investment. This comparative analysis also shows that PEC water splitting generates H<sub>2</sub> at one of the highest costs, averaging 6.9 \$ \$ kg<sub>H<sub>2</sub></sub><sup>-1</sup>. This result can be attributed to its early development stage, resulting in considerable expenses for producing and replacing PEs (which still have a low lifespan) and for manufacturing PEC panels (which still require the use of expensive and scarce materials, *i.e.* highly resistant polymers, noble metals).

After quantifying the environmental and economic performance of each H<sub>2</sub> production method, normalization was made using the following equation:<sup>351</sup>

$$\text{Ranking}(\text{method } i) = \frac{\max - \bar{p}_{x,i}}{\max - \min} \times 10 \quad (4)$$

where  $\bar{p}_{x,i}$  is the average value of parameter  $x$  of the H<sub>2</sub> production method  $i$ ; ranking equal to 0 translates to the poorest performance and 10 indicates the best performance (zero-cost and zero-emissions). The normalized results are presented in Fig. 14. PEC water splitting remains a promising option for sustainable H<sub>2</sub> production, scoring a remarkable environmental performance, but lacking economic arguments. It is important to note that the

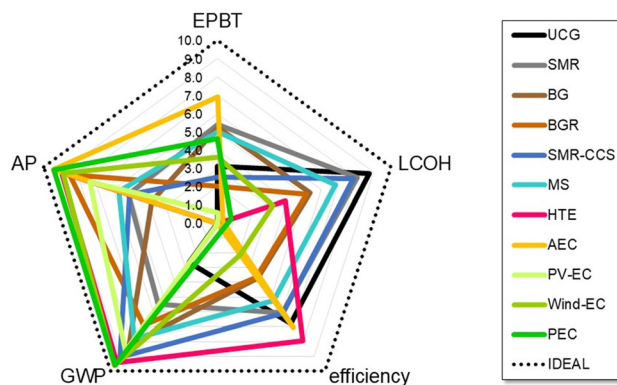


Fig. 14 Overall performance comparison of H<sub>2</sub> production processes.

use of biomass derivatives as feedstock for PEC-H<sub>2</sub> production can significantly improve the environmental performance comparatively to conventional PEC systems, as the oxidation of biomass shortens the pathway towards carbon neutrality and circular economy, also being quite interesting from an economic perspective.

On the other hand, more mature and widespread technologies, such as UCG and SMR, are financially safer, and more appealing to investors and H<sub>2</sub> consumers. So far, steam methane reforming with carbon capture and storage is the H<sub>2</sub> production technology that assures the best compromise between environment and



economics. Nevertheless, due to the geographically dispersed availability of primary renewable energy sources, *i.e.* wind, solar, and water resources, each case has to be analyzed separately, independently from the environmental and economic ranking; as an example, one can assume that wind-powered electrolysis would be a preferable option over PV-electrolysis for Nordic countries, such as Denmark, and solar-power electrolysis would be recommended for desert regions, such as North Africa or the Arabian peninsula. Overall, water electrolysis technologies offer more room for achieving a better environmental performance whereas fossil fuel-based technologies allow for a better economic performance.<sup>352,390</sup> However, the margin for reducing the environmental impact of fossil fuel-based technologies is very small, while the margin for improving the economic performance of renewable-based H<sub>2</sub> production processes is huge, which represents a vital endorsement for future investments in green-H<sub>2</sub> production.

## 7. Conclusions, closing remarks and future opportunities

PEC water splitting celebrated its 50th anniversary in 2022. Over 50 years, and side-by-side with a never-before-seen development in materials and technologies, PEC water splitting has been the subject of numerous publications and the forerunner of several discoveries and advancements in semiconductors, catalysts, nanostructures, and many others. Almost inexplicably, PEC-H<sub>2</sub> never left the lab and remains at a low TRL, still far from commercialization. Fortunately, the current global situation became a unique opportunity for the development of technologies that promote decarbonization, with a particular focus on the production of green fuels, where green-H<sub>2</sub> has clearly assumed the status of a hot topic. Indeed, the third decade of the 21st century witnessed sudden events that greatly affected the energy market dynamics, with a subsequent brutal impact on energy prices. World-leading countries started to question if the resulting economic turmoil could muffle their willingness to implement key changes in the energy market, endangering ambitious carbon neutrality goals, namely through green-H<sub>2</sub> production.<sup>393</sup> By 2022, fossil fuels supplied *ca.* 95% of the global H<sub>2</sub> production, more specifically 48% produced from natural gas, 30% from oil, and 18% from coal, which undermines the “green” labeling that H<sub>2</sub> is ambitioning.<sup>312,394</sup> Surprisingly, many private and government institutions interpreted the current situation as a great opportunity to implement drastic changes in the energy market, attributing to green-H<sub>2</sub> a prominent position in the financial packages to be distributed.<sup>395</sup> For example, in the European Green Deal, green-H<sub>2</sub> is a priority for achieving carbon neutrality by 2050. It is mentioned that the large-scale deployment of green-H<sub>2</sub> is mandatory for reducing GHG emissions towards 55% by 2030, requiring research and innovation efforts, manufacturing capabilities for supplying H<sub>2</sub> to large consumers, and especially, it is necessary to accelerate the development of promising technologies at low TRL, such as PEC water splitting.<sup>396,397</sup> For instance, within the European context, the latest update of the

Renewable Energy Directive (RED III) stipulates that the utilization of Renewable Fuels of Non-Biological Origin (RFNBOs), where renewable gaseous H<sub>2</sub> is included, will result in a carbon credit bonus for meeting the European targets mandated by the directive. As such, low-carbon fuels, particularly those whose production relies on green-H<sub>2</sub>, are currently witnessing unparalleled and distinctive valorisation, which is translated into a unique opportunity for PEC water splitting, as the technology is expected to have one of the lowest carbon footprints among H<sub>2</sub> production technologies. Sustainable energy production based on renewable sources and diversified technologies is essential to avoid sudden and steep fluctuations in energy prices due to unpredictable events, especially in countries where fossil fuel imports account for a considerable market share.<sup>398</sup>

There is no doubt that green-H<sub>2</sub> has gained the reputation of “key player” in future energy systems; still, a critical question remains: would PEC water splitting be able to compete with other green-H<sub>2</sub> production technologies? The simple answer is yes. Although in terms of thermodynamics and kinetics, the best PEC-H<sub>2</sub> systems can only hope to match the best PV-EC,<sup>399</sup> the differentiating competitive factor relies on the environmental and functional advantages of PEC water splitting, as PEC devices benefit from the integration of light absorbers with electrolysis technologies in a single unit, with no peripheral electronics.<sup>32</sup> LCA studies show that, among currently available H<sub>2</sub> production processes, PEC water splitting has the lowest environmental impact, despite coal gasification and steam methane reforming remaining the more financially appealing options to investors and H<sub>2</sub> consumers, due to their economy of scale and maturity. This reinforces that scalability is one of the most critical challenges for turning PEC water splitting economically viable in the upcoming years.<sup>15,57,277</sup> With state-of-the-art photoelectrodes, to reach a similar H<sub>2</sub> output as PV-EC, the active area of PEC devices must be at least 50 times larger, which means in the range of a few hundred centimeters.<sup>15</sup> Therefore, the commercial viability of PEC-H<sub>2</sub> production strongly relies on the successful upscaling of energy-efficient PES,<sup>183</sup> namely on achieving the benchmark values of the three-pillar criteria for commercialization, *i.e.* an active area of 100 cm<sup>2</sup>, 2920 h of stable performance (8 h over 365 days with a decrease of max. 10% of the initial current density) and a current density of 10 mA cm<sup>-2</sup>. Complementarily to the successful upscaling of efficient PES, using concentrated sunlight is a simple and clever way to boost H<sub>2</sub> production. For that, solar concentrators must be compatible with PEC device restrictions, and PEC devices must be compatible with the specifications of solar concentrators. No matter the concentrating solar technology, adopting thermal management strategies is essential to avoid performance losses due to the added thermal load. Finally, the competitiveness of PEC water splitting can be further enhanced using biomass derivatives, which allows overcoming thermodynamic and kinetic constraints of conventional systems and meeting circular economy requirements, by using wastewater as an electrolyte.

During the upcoming years, the field of PEC-H<sub>2</sub> production shall experience an unparalleled economic and research stimulus, which will allow for bridging lab-scale research with





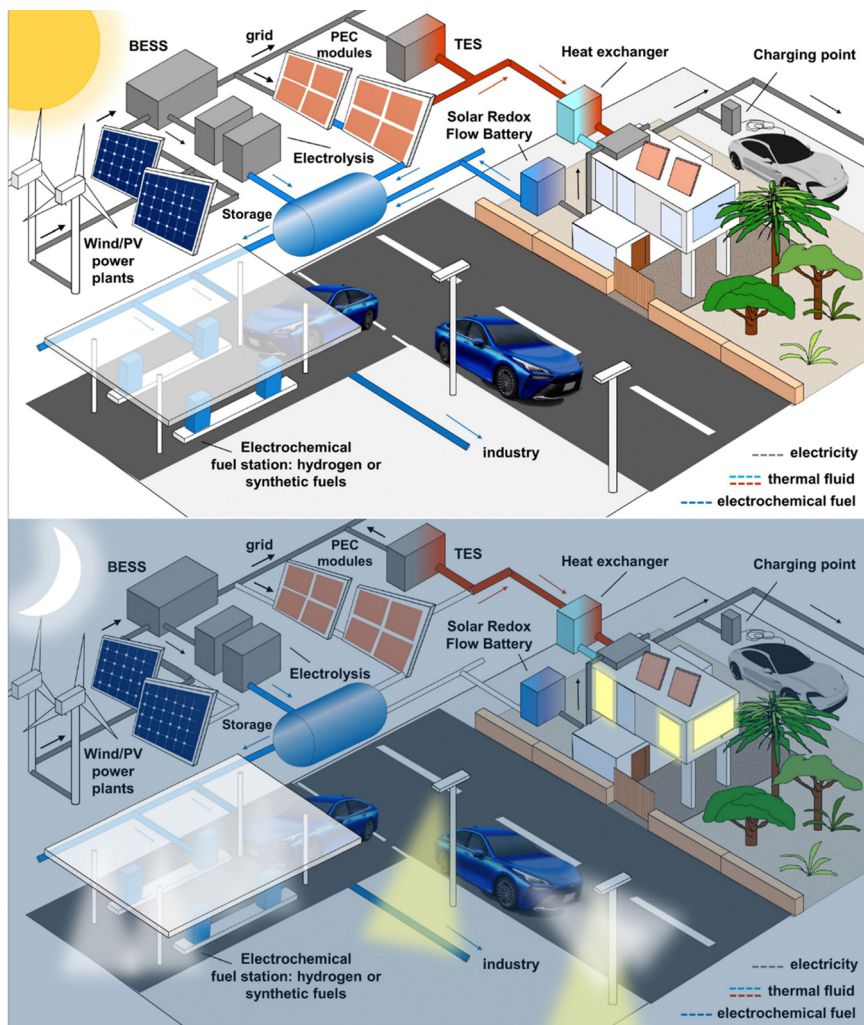


Fig. 15 Hypothetical scenario of the integration of PEC-H<sub>2</sub> production in the energy grid, comprising the electrical, thermal and fuel flows for day and night periods.

commercial interests. The best way forward is to accept, at first, that PEC devices will need to operate very close to their fundamental thermodynamic limits, at large scale.<sup>188,400</sup> This will require a solid convergence between academics and industry and research efforts should be directed to: (i) build and demonstrate large-scale PEC devices assembled in a highly recyclable embodiment, incorporating clever design features and CRM-free electrodes and catalysts; (ii) demonstrate high STH efficiency values for large-area PEs that are in the same range as the PV cells; (iii) develop preparation techniques for large-area PEs that can be easily implemented on an industrial scale; (iv) enhance the chemical stability of PEs targeting tens of thousands of hours; (v) investigate and develop new approaches for coupling solar concentrators with PEC devices; (vi) find suitable organic compounds for PEC-H<sub>2</sub> production systems, with the corresponding ideal PEs, catalysts and membranes; and (vii) reach consensual standardized methods for evaluating PEC devices, based on environmental and techno-economic metrics.

If these targets are not met in the next decade, the feasibility of PEC-H<sub>2</sub> production will be seriously compromised and the

topic could tend to become another subject without practical application for society. Nevertheless, investment in research on PEC devices should move forward, especially given the quest for decarbonization. Moreover, PEC devices are finding other applications beyond water splitting, such as in solar redox flow cells.<sup>68,401</sup> Overall, the incorporation of PEC-H<sub>2</sub> production into the energy market should be a gradual process and should not aim at dethroning existing technologies, but aspire instead to be part of a symbiotic integration (Fig. 15).

## Abbreviation

AE	Auxiliary electrode
AEC	Alkaline electrolysis
AEM	Anion exchange membrane
ALD	Atomic layer deposition
AP	Acidification potential
ATO	Antimony-doped tin oxide
AZO	Aluminium-doped zinc oxide



BESS	Battery energy storage system	PSC	Perovskite solar cell
BG	Biomass gasification	PTC	Parabolic trough collector
BGR	Biogas reforming	PV	Photovoltaic
BHJ	Bulk heterojunction	PV-EC	Photovoltaic-electrolysis
BIPV	Building-integrated photovoltaic	QD	Quantum dot
BOS	Balance of system	RE	Reference electrode
BPM	Bipolar membrane	RFNBO	Renewable fuel from non-biological origin
CAPEX	Capital Expenditure	RHE	Reversible hydrogen electrode
CCM	Catalyst-coated membranes	SMR	Steam methane reforming
CCS	Carbon capture and storage	SOEC	Solid oxide electrolyser cell
CE	Counter-electrode	SP	Spray pyrolysis
CEM	Cation exchange membrane	STH	Solar-to-hydrogen
CIPEC	Concentrator-integrated photoelectrochemical	TCO	Transparent conductive oxide
CL	Convex lenses	TE	Thermoelectric
CPC	Compound parabolic concentrator	TES	Thermal energy storage
CPEC	Concentrated-photoelectrochemical	TEG	Thermoelectric generator
CPVE	Concentrated photovoltaic electrolysis	TRL	Technological readiness level
CRM	Critical raw material	UPORTO	University of Porto
CZTS	Copper zinc tin sulfide	UCG	Underground coal gasification
DLR	German Aerospace Centre (Deutsches Zentrum für Luft- und Raumfahrt)	US	United States
DSSC	Dye-sensitized solar cell	UV	Ultraviolet
EC	Electrolysis/electrolyser	WE	Working electrode
EIS	Electrochemical impedance spectroscopy	WS	Working sense
EPBT	Energy payback time		
ERoEI	Energy return on energy invested		
ETCO	Embedded transparent conductive oxide		
EU	European Union		
FL	Fresnel lenses		
FTO	Fluorine-doped tin oxide		
FOM	Figure-of-merit		
GHG	Greenhouse gases		
GLAD	Glancing angle deposition		
GWP	Global warming potential		
HER	Hydrogen evolution reaction		
HTE	High-temperature electrolysis		
IPEC	Integrated photoelectrochemical		
ITO	Indium tin oxide		
LCA	Life cycle assessment		
LCOH	Levelized cost of hydrogen		
LFR	Linear Fresnel reflector		
LSC	Luminescent solar concentrator		
MEA	Membrane electrode assembly		
MOFs	Metal-organic framework-based materials		
MS	Methane splitting		
OER	Oxygen evolution reaction		
PCE	Power conversion efficiency		
PDC	Parabolic dish concentrator		
PE	Photoelectrode		
PEC	Photoelectrochemical		
PEEK	Polyether ether ketone		
PEM	Polymer electrolyte membrane		
PET	Polyethylene		
PFSA	Perfluorinated sulfonic acid		
PMMA	Poly(methyl 2-methylpropenoate)		
PS	Polystyrene		

## Conflicts of interest

There are no conflicts to declare.

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## References

- C. S. Lai, Y. Jia, L. L. Lai, Z. Xu, M. D. McCulloch and K. P. Wong, *Renewable Sustainable Energy Rev.*, 2017, **78**, 439–451.
- N. Abas, A. Kalair and N. Khan, *Futures*, 2015, **69**, 31–49.



- 3 I. Mesquita, L. Andrade and A. Mendes, *Renewable Sustainable Energy Rev.*, 2018, **82**, 2471–2489.
- 4 M. González-Eguino, *Renewable Sustainable Energy Rev.*, 2015, **47**, 377–385.
- 5 M. A. Khan, M. Z. Khan, K. Zaman and L. Naz, *Renewable Sustainable Energy Rev.*, 2014, **29**, 336–344.
- 6 A. Vaona, *Renewable Energy*, 2016, **86**, 354–359.
- 7 K. Zaman and M. Abd-el Moemen, *Renewable Sustainable Energy Rev.*, 2017, **74**, 1119–1130.
- 8 G. Nicoletti, N. Arcuri, G. Nicoletti and R. Bruno, *Energy Convers. Manage.*, 2015, **89**, 205–213.
- 9 G. P. Peters, R. M. Andrew, J. G. Canadell, S. Fuss, R. B. Jackson, J. I. Korsbakken, C. Le Quéré and N. Nakicenovic, *Nat. Clim. Change*, 2017, **2**, 118–122.
- 10 H. J. Schellnhuber, S. Rahmstorf and R. Winkelmann, *Nat. Clim. Change*, 2016, **6**, 649–653.
- 11 M. Lee, J. Yun, A. Pyka, D. Won, F. Kodama, G. Schiuma, H. Park, J. Jeon, K. Park and K. Jung, *J. Open Innov.: Technol. Mark. Complex.*, 2018, **4**, 21.
- 12 F. Yilmaz, M. T. Balta and R. Selbaş, *Renewable Sustainable Energy Rev.*, 2016, **56**, 171–178.
- 13 F. Zhang, P. Zhao, M. Niu and J. Maddy, *Int. J. Hydrogen Energy*, 2016, **41**, 14535–14552.
- 14 F. Suleman, I. Dincer and M. Agelin-Chaab, *Int. J. Hydrogen Energy*, 2015, **40**, 6976–6987.
- 15 S. Ardo, D. F. Rivas, M. A. Modestino, V. S. Greiving, F. F. Abdi, E. A. Llado, V. Artero, K. Ayers, C. Battaglia and J.-P. Becker, *Energy Environ. Sci.*, 2018, **11**, 2768–2783.
- 16 W. Yang, R. R. Prabhakar, J. Tan, S. D. Tilley and J. Moon, *Chem. Soc. Rev.*, 2019, **48**, 4979–5015.
- 17 S. Tembhurne, F. Nandjou and S. Haussener, *Nat. Energy*, 2019, **4**, 399.
- 18 D. M. Fabian, S. Hu, N. Singh, F. A. Houle, T. Hisatomi, K. Domen, F. E. Osterloh and S. Ardo, *Energy Environ. Sci.*, 2015, **8**, 2825–2850.
- 19 Y. Goto, T. Hisatomi, Q. Wang, T. Higashi, K. Ishikiriya, T. Maeda, Y. Sakata, S. Okunaka, H. Tokudome and M. Katayama, *Joule*, 2018, **2**, 509–520.
- 20 T. Takata and K. Domen, *ACS Energy Lett.*, 2019, **4**, 542–549.
- 21 B. Turan, J.-P. Becker, F. Urbain, F. Finger, U. Rau and S. Haas, *Nat. Commun.*, 2016, **7**, 12681.
- 22 W. H. Brattain and J. Bardeen, *Bell Syst. Tech. J.*, 1953, **32**, 1–41.
- 23 C. Garrett and W. H. Brattain, *Phys. Rev.*, 1955, **99**, 376.
- 24 H. Gerischer, *J. Electrochem. Soc.*, 1966, **113**, 1174.
- 25 P. Boddy, *J. Electrochem. Soc.*, 1968, **115**, 199.
- 26 A. Fujishima and K. Honda, *Nature*, 1972, **238**, 37–38.
- 27 EPFL, SolarFuelsDB, <https://solarfuelsdb.epfl.ch/>, (accessed July, 2023).
- 28 K. T. Fountaine, H. J. Lewerenz and H. A. Atwater, *Nat. Commun.*, 2016, **7**, 13706.
- 29 Technical Targets for Hydrogen Production from Photoelectrochemical Water Splitting, <https://www.energy.gov/eere/fuelcells/doe-technical-targets-hydrogen-production-photoelectrochemical-water-splitting>, (accessed July 2023, 2023).
- 30 W.-H. Cheng, M. H. Richter, M. M. May, J. Ohlmann, D. Lackner, F. Dimroth, T. Hannappel, H. A. Atwater and H.-J. Lewerenz, *ACS Energy Lett.*, 2018, **3**, 1795–1800.
- 31 S. Hu, C. Xiang, S. Haussener, A. D. Berger and N. S. Lewis, *Energy Environ. Sci.*, 2013, **6**, 2984–2993.
- 32 A. Rothschild and H. Dotan, *ACS Energy Lett.*, 2017, **2**, 45–51.
- 33 J. H. Kim, J.-W. Jang, Y. H. Jo, F. F. Abdi, Y. H. Lee, R. van de Krol and J. S. Lee, *Nat. Commun.*, 2016, **7**, 13380.
- 34 X. Shi, H. Jeong, S. J. Oh, M. Ma, K. Zhang, J. Kwon, I. T. Choi, I. Y. Choi, H. K. Kim, J. K. Kim and J. H. Park, *Nat. Commun.*, 2016, **7**, 11943.
- 35 L. Pan, J. H. Kim, M. T. Mayer, M.-K. Son, A. Ummadisingu, J. S. Lee, A. Hagfeldt, J. Luo and M. Grätzel, *Nat. Catal.*, 2018, **1**, 412–420.
- 36 S. K. Karuturi, H. Shen, A. Sharma, F. J. Beck, P. Varadhan, T. Duong, P. R. Narangari, D. Zhang, Y. Wan, J.-H. He, H. H. Tan, C. Jagadish and K. Catchpole, *Adv. Energy Mater.*, 2020, **10**, 2000772.
- 37 B. A. Pinaud, J. D. Benck, L. C. Seitz, A. J. Forman, Z. Chen, T. G. Deutsch, B. D. James, K. N. Baum, G. N. Baum and S. Ardo, *Energy Environ. Sci.*, 2013, **6**, 1983–2002.
- 38 M. R. Shaner, H. A. Atwater, N. S. Lewis and E. W. McFarland, *Energy Environ. Sci.*, 2016, **9**, 2354–2371.
- 39 T. Lopes, P. Dias, L. Andrade and A. Mendes, *Sol. Energy Mater. Sol. Cells*, 2014, **128**, 399–410.
- 40 S. Haussener, C. Xiang, J. M. Spurgeon, S. Ardo, N. S. Lewis and A. Z. Weber, *Energy Environ. Sci.*, 2012, **5**, 9922–9935.
- 41 I. Holmes-Gentle, F. Hoffmann, C. A. Mesa and K. Hellgardt, *Sustainable Energy Fuels*, 2017, **1**, 1184–1198.
- 42 M. T. Spitler, M. A. Modestino, T. G. Deutsch, C. X. Xiang, J. R. Durrant, D. V. Esposito, S. Haussener, S. Maldonado, I. D. Sharp, B. A. Parkinson, D. S. Ginley, F. A. Houle, T. Hannappel, N. R. Neale, D. G. Nocera and P. C. McIntyre, *Sustainable Energy Fuels*, 2020, **4**, 985–995.
- 43 A. Vilanova, T. Lopes, C. Spenke, M. Wullenkord and A. Mendes, *Energy Storage Mater.*, 2018, **13**, 175–188.
- 44 R. van de Krol, in *Photoelectrochemical Hydrogen Production*, ed. R. van de Krol and M. Grätzel, Springer Science + Business Media, New York, USA, 2012, vol. VIII, ch. 2, pp. 13–67.
- 45 X. Li, J. Yu, J. Low, Y. Fang, J. Xiao and X. Chen, *J. Mater. Chem. A*, 2015, **3**, 2485–2534.
- 46 N. Papageorgiou, *Coord. Chem. Rev.*, 2004, **248**, 1421–1446.
- 47 F. Le Formal, *On the Morphology and Interfaces of Nanostructured Hematite Photoanodes for Solar-Driven Water Splitting*, PhD thesis, EPFL, Lausanne, Switzerland, 2011, p. 216, DOI: [10.5075/epfl-thesis-5180](https://doi.org/10.5075/epfl-thesis-5180).
- 48 H. Dotan, N. Mathews, T. Hisatomi, M. Grätzel and A. Rothschild, *J. Phys. Chem. Lett.*, 2014, **5**, 3330–3334.
- 49 C. Ilkay, *Solar photoelectrolysis of water with translucent nano-structured hematite photoanodes*, PhD thesis, EPFL, Lausanne, Switzerland, 2007, p. 200, DOI: [10.5075/epfl-thesis-3914](https://doi.org/10.5075/epfl-thesis-3914).
- 50 A. G. Tamirat, J. Rick, A. A. Dubale, W.-N. Su and B.-J. Hwang, *Nanoscale Horiz.*, 2016, **1**, 243–267.





- 51 P. Zhai, S. Haussener, J. Ager, R. Sathre, K. Walczak, J. Greenblatt and T. McKone, *Energy Environ. Sci.*, 2013, **6**, 2380–2389.
- 52 T. Hisatomi, J. Kubota and K. Domen, *Chem. Soc. Rev.*, 2014, **43**, 7520–7535.
- 53 F. A. S. Moranchell, J. M. S. Pineda, J. N. H. Pérez, U. S. Silva-Rivera, C. A. C. Escobedo and R. d. G. G. Huerta, *Int. J. Hydrogen Energy*, 2020, **45**, 13683–13692.
- 54 K. Zeng and D. Zhang, *Prog. Energy Combust. Sci.*, 2010, **36**, 307–326.
- 55 C. Jiang, S. J. Moniz, A. Wang, T. Zhang and J. Tang, *Chem. Soc. Rev.*, 2017, **46**, 4645–4660.
- 56 K. Sivula and R. Van De Krol, *Nat. Rev. Mater.*, 2016, **1**, 15010.
- 57 J. H. Kim, D. Hansora, P. Sharma, J.-W. Jang and J. S. Lee, *Chem. Soc. Rev.*, 2019, **48**, 1908–1971.
- 58 S. K. Saraswat, D. D. Rodene and R. B. Gupta, *Renewable Sustainable Energy Rev.*, 2018, **89**, 228–248.
- 59 S. Shen, S. A. Lindley, X. Chen and J. Z. Zhang, *Energy Environ. Sci.*, 2016, **9**, 2744–2775.
- 60 D. Wang, X.-T. Zhang, P.-P. Sun, S. Lu, L.-L. Wang, Y.-A. Wei and Y.-C. Liu, *Int. J. Hydrogen Energy*, 2014, **39**(28), 16212–16219.
- 61 M. M. Momeni and Y. Ghayeb, *J. Alloys Compd.*, 2015, **637**, 393–400.
- 62 X. Li, P. S. Bassi, P. P. Boix, Y. Fang and L. H. Wong, *ACS Appl. Mater. Interfaces*, 2015, **7**, 16960–16966.
- 63 C. Ros, T. Andreu and J. R. Morante, *J. Mater. Chem. A*, 2020, **8**, 10625–10669.
- 64 P. Dias, A. Vilanova, T. Lopes, L. Andrade and A. Mendes, *Nano Energy*, 2016, **23**, 70–79.
- 65 I. S. Cho, H. S. Han, M. Logar, J. Park and X. Zheng, *Adv. Energy Mater.*, 2016, **6**, 1501840.
- 66 K. Sivula, F. Le Formal and M. Gratzel, *ChemSusChem*, 2011, **4**, 432–449.
- 67 T. H. Jeon, G.-h Moon, H. Park and W. Choi, *Nano Energy*, 2017, **39**, 211–218.
- 68 T. da Silva Lopes, P. Dias, R. Monteiro, A. Vilanova, D. Ivanou and A. Mendes, *Adv. Energy Mater.*, 2021, 2102893.
- 69 A. Landman, R. Halabi, P. Dias, H. Dotan, A. Mehlmann, G. E. Shter, M. Halabi, O. Naseraldeem, A. Mendes, G. S. Grader and A. Rothschild, *Joule*, 2020, **4**, 448–471.
- 70 A. Apolinário, T. Lopes, C. D. Costa, J. P. Araújo and A. Mendes, *ACS Appl. Energy Mater.*, 2019, **2**, 1040–1050.
- 71 Y. Wang, W. Tian, C. Chen, W. Xu and L. Li, *Adv. Funct. Mater.*, 2019, **29**, 1809036.
- 72 T. Zhu, M. N. Chong and E. S. Chan, *ChemSusChem*, 2014, **7**, 2974–2997.
- 73 A. Tacca, L. Meda, G. Marra, A. Savoini, S. Caramori, V. Cristino, C. A. Bignozzi, V. G. Pedro, P. P. Boix and S. Gimenez, *Chem. Phys. Chem.*, 2012, **13**, 3025–3034.
- 74 W. Kwong, H. Qiu, A. Nakaruk, P. Koshy and C. Sorrell, *Energy Procedia*, 2013, **34**, 617–626.
- 75 S. Reinhard, F. Rechberger and M. Niederberger, *Chem-PlusChem*, 2016, **81**, 935.
- 76 Y. Pihosh, I. Turkevych, K. Mawatari, J. Uemura, Y. Kazoe, S. Kosar, K. Makita, T. Sugaya, T. Matsui and D. Fujita, *Sci. Rep.*, 2015, **5**, 11141.
- 77 F. F. Abdi, L. Han, A. H. Smets, M. Zeman, B. Dam and R. van de Krol, *Nat. Commun.*, 2013, **4**, 2195.
- 78 F. F. Abdi, L. Han, A. H. M. Smets, M. Zeman, B. Dam and R. van de Krol, *Nat. Commun.*, 2013, **4**, 2195.
- 79 I. Y. Ahmet, Y. Ma, J.-W. Jang, T. Henschel, B. Stannowski, T. Lopes, A. Vilanova, A. Mendes, F. Abdi and R. van de Krol, *Sustainable Energy Fuels*, 2019, **3**, 2366–2379.
- 80 K. Ouyang, S. Xie, P. Wang, J. Zhu and P. Zhan, *Int. J. Hydrogen Energy*, 2019, **44**, 7288–7299.
- 81 M. Zhou, J. Bao, Y. Xu, J. Zhang, J. Xie, M. Guan, C. Wang, L. Wen, Y. Lei and Y. Xie, *ACS Nano*, 2014, **8**, 7088–7098.
- 82 S. Kosar, Y. Pihosh, I. Turkevych, K. Mawatari, J. Uemura, Y. Kazoe, K. Makita, T. Sugaya, T. Matsui and D. Fujita, *Jpn. J. Appl. Phys.*, 2016, **55**, 04ES01.
- 83 Y. Kuang, Q. Jia, G. Ma, T. Hisatomi, T. Minegishi, H. Nishiyama, M. Nakabayashi, N. Shibata, T. Yamada and A. Kudo, *Nat. Energy*, 2016, **2**, 1–9.
- 84 G. Liu, S. Ye, P. Yan, F. Xiong, P. Fu, Z. Wang, Z. Chen, J. Shi and C. Li, *Energy Environ. Sci.*, 2016, **9**, 1327–1334.
- 85 G. Liu, J. Shi, F. Zhang, Z. Chen, J. Han, C. Ding, S. Chen, Z. Wang, H. Han and C. Li, *Angew. Chem., Int. Ed.*, 2014, **53**, 7295–7299.
- 86 A. Ishikawa, T. Takata, J. N. Kondo, M. Hara and K. Domen, *J. Phys. Chem. B*, 2004, **108**, 11049–11053.
- 87 M. Higashi, K. Domen and R. Abe, *Energy Environ. Sci.*, 2011, **4**, 4138–4147.
- 88 T. Higashi, H. Nishiyama, V. Nandal, Y. Pihosh, Y. Kawase, R. Shoji, M. Nakabayashi, Y. Sasaki, N. Shibata and H. Matsuzaki, *Energy Environ. Sci.*, 2022, **15**, 4761–4775.
- 89 Y.-F. Xu, H.-S. Rao, X.-D. Wang, H.-Y. Chen, D.-B. Kuang and C.-Y. Su, *J. Mater. Chem. A*, 2016, **4**, 5124–5129.
- 90 D. H. Taffa, R. Dillert, A. C. Ulpe, K. C. Bauerfeind, T. Bredow, D. W. Bahnemann and M. Wark, *J. Photonics Energy*, 2016, **7**, 012009.
- 91 J. Azevedo, S. D. Tilley, M. Schreier, M. Stefik, C. Sousa, J. P. Araújo, A. Mendes, M. Grätzel and M. T. Mayer, *Nano Energy*, 2016, **24**, 10–16.
- 92 J. Azevedo, L. Steier, P. Dias, M. Stefik, C. Sousa, J. Araujo, A. Mendes, M. Graetzel and S. Tilley, *Energy Environ. Sci.*, 2014, **7**, 4044–4052.
- 93 S. D. Tilley, M. Schreier, J. Azevedo, M. Stefik and M. Graetzel, *Adv. Funct. Mater.*, 2013, **24**, 1616–3028.
- 94 D. Bae, B. Seger, P. C. Vesborg, O. Hansen and I. Chorkendorff, *Chem. Soc. Rev.*, 2017, **46**, 1933–1954.
- 95 Y. Lin, C. Battaglia, M. Boccard, M. Hettick, Z. Yu, C. Ballif, J. W. Ager and A. Javey, *Nano Lett.*, 2013, **13**, 5615–5618.
- 96 H.-C. Fu, P. Varadhan, M.-L. Tsai, W. Li, Q. Ding, C.-H. Lin, M. Bonifazi, A. Fratolocchi, S. Jin and J.-H. He, *Nano Energy*, 2020, **70**, 104478.
- 97 A. Vilanova, P. Dias, J. Azevedo, M. Wullenkord, C. Spence, T. Lopes and A. Mendes, *J. Power Sources*, 2020, **454**, 227890.
- 98 D. K. Lee and K.-S. Choi, *Nat. Energy*, 2018, **3**, 53–60.
- 99 Z. Zhang, X. Huang, B. Zhang and Y. Bi, *Energy Environ. Sci.*, 2022, **15**, 2867–2873.
- 100 W. J. Lee, P. S. Shinde, G. H. Go and E. Ramasamy, *Int. J. Hydrogen Energy*, 2011, **36**, 5262–5270.





- 101 K. Jakubow-Piotrowska, D. Kurzydowski, P. Wrobel and J. Augustynski, *ACS Phys. Chem. Au*, 2022, **2**, 299–304.
- 102 Final Report Summary – PECDEMO (Photoelectrochemical Demonstrator Device for Solar Hydrogen Generation), 2017. Available online: <https://cordis.europa.eu/project/id/621252/reporting> (accessed on 3 August 2023).
- 103 A. Gonçalves, *Development of Tantalum Nitride Photoelectrodes*, PhD thesis, UPORTO-FEUP, Porto, Portugal, 2022, p. 195, <https://hdl.handle.net/10216/147462>.
- 104 F. Santos, J. Martins, J. Capitão, S. Emami, D. Ivanou and A. Mendes, *ACS Appl. Energy Mater.*, 2022, **5**, 7220–7229.
- 105 A. Sekar and K. Sivula, *Chimia*, 2021, **75**, 169.
- 106 L. Steier and S. Holliday, *J. Mater. Chem. A*, 2018, **6**, 21809–21826.
- 107 S. Otep, T. Michinobu and Q. Zhang, *Sol. RRL*, 2020, **4**, 1900395.
- 108 Q. Yin, K. Zhang, L. Zhang, J. Jia, X. Zhang, S. Pang, Q.-H. Xu, C. Duan, F. Huang and Y. Cao, *J. Mater. Chem. A*, 2019, **7**, 12426–12433.
- 109 T. Ameri, G. Dennler, C. Lungenschmied and C. J. Brabec, *Energy Environ. Sci.*, 2009, **2**, 347–363.
- 110 L. Yao, A. Rahmanudin, N. Guijarro and K. Sivula, *Adv. Energy Mater.*, 2018, **8**, 1802585.
- 111 J. M. Yu, J. Lee, Y. S. Kim, J. Song, J. Oh, S. M. Lee, M. Jeong, Y. Kim, J. H. Kwak and S. Cho, *Nat. Commun.*, 2020, **11**, 5509.
- 112 D. Zhang, H. H. Cho, J. H. Yum, M. Mensi and K. Sivula, *Adv. Energy Mater.*, 2022, **12**, 2202363.
- 113 X. Li, Z. Wang and L. Wang, *Small Science*, 2021, **1**, 2000074.
- 114 J. B. Pan, B. H. Wang, J. B. Wang, H. Z. Ding, W. Zhou, X. Liu, J. R. Zhang, S. Shen, J. K. Guo and L. Chen, *Angew. Chem., Int. Ed.*, 2021, **60**, 1433–1440.
- 115 A. S. Bati, Y. L. Zhong, P. L. Burn, M. K. Nazeeruddin, P. E. Shaw and M. Batmunkh, *Commun. Mater.*, 2023, **4**, 2.
- 116 H. Min, D. Y. Lee, J. Kim, G. Kim, K. S. Lee, J. Kim, M. J. Paik, Y. K. Kim, K. S. Kim and M. G. Kim, *Nature*, 2021, **598**, 444–450.
- 117 W. Wang, M. Xu, X. Xu, W. Zhou and Z. Shao, *Angew. Chem., Int. Ed.*, 2020, **59**, 136–152.
- 118 S. Bhattacharjee, V. Andrei, C. Pornrungroj, M. Rahaman, C. M. Pichler and E. Reisner, *Adv. Funct. Mater.*, 2022, **32**, 2109313.
- 119 E. Edwardes Moore, V. Andrei, S. Zacarias, I. A. Pereira and E. Reisner, *ACS Energy Lett.*, 2019, **5**, 232–237.
- 120 V. Andrei, G. M. Ucoski, C. Pornrungroj, C. Uswachoke, Q. Wang, D. S. Achilleos, H. Kasap, K. P. Sokol, R. A. Jagt and H. Lu, *Nature*, 2022, **608**, 518–522.
- 121 A. Guerrero and J. Bisquert, *Curr. Opin. Electrochem.*, 2017, **2**, 144–147.
- 122 G. Liu, S. K. Karuturi, H. Chen, D. Wang, J. W. Ager, A. N. Simonov and A. Tricoli, *Sol. Energy*, 2020, **202**, 198–203.
- 123 H. Chen, M. Zhang, T. Tran-Phu, R. Bo, L. Shi, I. Di Bernardo, J. Bing, J. Pan, S. Singh and J. Lipton-Duffin, *Adv. Funct. Mater.*, 2021, **31**, 2008245.
- 124 M. Daboczi, J. Cui, F. Temerov and S. Eslava, 2023.
- 125 S. Oueslati, M. Pilvet, M. Grossberg, M. Kauk-Kuusik, J. Krustok and D. Meissner, *Thin Solid Films*, 2021, **739**, 138981.
- 126 D. Yokoyama, T. Minegishi, K. Jimbo, T. Hisatomi, G. Ma, M. Katayama, J. Kubota, H. Katagiri and K. Domen, *Appl. Phys. Express*, 2010, **3**, 101202.
- 127 C. Ros, T. Andreu, S. Giraldo, V. Izquierdo-Roca, E. Saucedo and J. R. Morante, *ACS Appl. Mater. Interfaces*, 2018, **10**, 13425–13433.
- 128 M. Vishwakarma, M. Kumar, M. Hendrickx, J. Hadermann, A. P. Singh, Y. Batra and B. Mehta, *Adv. Mater. Interfaces*, 2021, **8**, 2002124.
- 129 Y. Park, K. J. McDonald and K.-S. Choi, *Chem. Soc. Rev.*, 2013, **42**, 2321–2337.
- 130 P. Dias and A. Mendes, in *Encyclopedia of Sustainability Science and Technology*, ed. R. A. Meyers, Springer, New York, New York, NY, 2017, pp.1–52, DOI: [10.1007/978-1-4939-2493-6\\_957-1](https://doi.org/10.1007/978-1-4939-2493-6_957-1).
- 131 J. Zhou, S. Lin, Y. Chen and A. Gaskov, *Appl. Surf. Sci.*, 2017, **403**, 274–281.
- 132 Y. Kuang, Q. Jia, H. Nishiyama, T. Yamada, A. Kudo and K. Domen, *Adv. Energy Mater.*, 2016, **6**, 1501645.
- 133 F. Le Formal, M. Grätzel and K. Sivula, *Adv. Funct. Mater.*, 2010, **20**, 1099–1107.
- 134 X. Zhao, J. Feng, S. Chen, Y. Huang, T. C. Sum and Z. Chen, *Phys. Chem. Chem. Phys.*, 2017, **19**, 1074–1082.
- 135 M. J. Katz, S. C. Riha, N. C. Jeong, A. B. Martinson, O. K. Farha and J. T. Hupp, *Coord. Chem. Rev.*, 2012, **256**, 2521–2529.
- 136 Y. Ling, G. Wang, D. A. Wheeler, J. Z. Zhang and Y. Li, *Nano Lett.*, 2011, **11**, 2119–2125.
- 137 I. Cesar, K. Sivula, A. Kay, R. Zboril and M. Grätzel, *J. Phys. Chem. C*, 2008, **113**, 772–782.
- 138 Y.-S. Hu, A. Kleiman-Shwarsctein, A. J. Forman, D. Hazen, J.-N. Park and E. W. McFarland, *Chem. Mater.*, 2008, **20**, 3803–3805.
- 139 J. H. Kim and J. S. Lee, *Adv. Mater.*, 2019, **31**, 1806938.
- 140 P. Dias, L. Andrade and A. Mendes, *Nano Energy*, 2017, **38**, 218–231.
- 141 M. Li, Y. Yang, Y. Ling, W. Qiu, F. Wang, T. Liu, Y. Song, X. Liu, P. Fang and Y. Tong, *Nano Lett.*, 2017, **17**, 2490–2495.
- 142 K. D. Malviya, H. Dotan, D. Shlenkevich, A. Tsyganok, H. Mor and A. Rothschild, *J. Mater. Chem. A*, 2016, **4**, 3091–3099.
- 143 D. Cao, W. Luo, J. Feng, X. Zhao, Z. Li and Z. Zou, *Energy Environ. Sci.*, 2014, **7**, 752–759.
- 144 M. R. Nellist, F. A. Laskowski, F. Lin, T. J. Mills and S. W. Boettcher, *Acc. Chem. Res.*, 2016, **49**, 733–740.
- 145 X. Zou and Y. Zhang, *Chem. Soc. Rev.*, 2015, **44**, 5148–5180.
- 146 T. R. Hellstern, J. D. Benck, J. Kibsgaard, C. Hahn and T. F. Jaramillo, *Adv. Energy Mater.*, 2016, **6**, 1501758.
- 147 H. Gerischer, *J. Electroanal. Chem. Interfacial Electrochem.*, 1977, **82**, 133–143.
- 148 B. Han, M. Risch, S. Belden, S. Lee, D. Bayer, E. Mutoro and Y. Shao-Horn, *J. Electrochem. Soc.*, 2018, **165**, F813.
- 149 L. Alibabaei, M. K. Brennaman, M. R. Norris, B. Kalanyan, W. Song, M. D. Losego, J. J. Concepcion, R. A. Binstead, G. N. Parsons and T. J. Meyer, *Proc. Natl. Acad. Sci. U. S. A.*, 2013, **110**, 20008–20013.



- 150 H. S. Oh, H. N. Nong and P. Strasser, *Adv. Funct. Mater.*, 2015, **25**, 1074–1081.
- 151 J. Deng, Q. Zhang, K. Feng, H. Lan, J. Zhong, M. Chaker and D. Ma, *ChemSusChem*, 2018, **11**, 3783–3789.
- 152 K. J. McDonald and K.-S. Choi, *Chem. Mater.*, 2011, **23**, 1686–1693.
- 153 M. Zhang, R. P. Antony, S. Y. Chiam, F. F. Abdi and L. H. Wong, *ChemSusChem*, 2019, **12**, 2022–2028.
- 154 S. Hu, N. S. Lewis, J. W. Ager, J. Yang, J. R. McKone and N. C. Strandwitz, *J. Phys. Chem. C*, 2015, **119**, 24201–24228.
- 155 T. Lopes, L. Andrade and A. Mendes, *Sol. Energy Sci. Eng. Appl.*, 2013, **128**, 399–410.
- 156 C. Xiang, A. Z. Weber, S. Ardo, A. Berger, Y. Chen, R. Coridan, K. T. Fountaine, S. Haussener, S. Hu and R. Liu, *Angew. Chem., Int. Ed.*, 2016, **55**, 12974–12988.
- 157 R. Memming, *Semiconductor electrochemistry*, John Wiley & Sons, Germany, 2nd edn, 2015, p. 484.
- 158 M. M. Momeni, M. Mahvari and Y. Ghayeb, *J. Electroanal. Chem.*, 2019, **832**, 7–23.
- 159 X. Lv, I. Rodriguez, C. Hu, J. Shang, P. H.-L. Sit, C. Ye, G. Oskam and W. Y. Teoh, *Mater. Today Chem.*, 2019, **12**, 7–15.
- 160 P. Dias, T. Lopes, L. Meda, L. Andrade and A. Mendes, *Phys. Chem. Chem. Phys.*, 2016, **18**, 5232–5243.
- 161 S. D. Ponja, B. A. Williamson, S. Sathasivam, D. O. Scanlon, I. P. Parkin and C. J. Carmalt, *J. Mater. Chem. C*, 2018, **6**, 7257–7266.
- 162 N. Noor and I. P. Parkin, *J. Mater. Chem. C*, 2013, **1**, 984–996.
- 163 R. M. Pasquarelli, D. S. Ginley and R. O'Hayre, *Chem. Soc. Rev.*, 2011, **40**, 5406–5441.
- 164 D.-w. Choi, W. Maeng and J.-S. Park, *Appl. Surf. Sci.*, 2014, **313**, 585–590.
- 165 E. M. Bomhard, *Environ. Toxicol. Pharmacol.*, 2018, **58**, 250–258.
- 166 C. Khelifi, A. Attaf, A. Yahia and M. Dahnoun, *Surf. Interfaces*, 2019, **15**, 244–249.
- 167 A. Way, J. Luke, A. D. Evans, Z. Li, J.-S. Kim, J. R. Durrant, H. K. Hin Lee and W. C. Tsoi, *AIP Adv.*, 2019, **9**, 085220.
- 168 M. Asadzadeh, F. Tajabadi, D. Dastan, P. Sangpour, Z. Shi and N. Taghavinia, *Ceram. Int.*, 2021, **47**, 5487–5494.
- 169 I. Y. Bu, *Ceram. Int.*, 2014, **40**, 417–422.
- 170 A. Kumar, S. K. Swami and V. Dutta, *J. Alloys Compd.*, 2014, **588**, 546–550.
- 171 J. T. Wang, X. L. Shi, W. W. Liu, X. H. Zhong, J. N. Wang, L. Pyrah, K. D. Sanderson, P. M. Ramsey, M. Hirata and K. Tsuru, *Sci. Rep.*, 2014, **4**, 1–9.
- 172 X. Pinheiro, A. Vilanova, D. Mesquita, M. Monteiro, J. Eriksson, J. Barbosa, C. Matos, A. Oliveira, K. Oliveira and J. Capitão, *Ceram. Int.*, 2022, **49**(8), 13019–13030.
- 173 T.-F. Hou, A. Shanmugasundaram, I. V. Bagal, S.-W. Ryu and D.-W. Lee, *Mater. Sci. Semicond. Process.*, 2021, **121**, 105445.
- 174 A. Thakur, P. Kumar, S. M. Thalluri, R. Sinha and P. Devi, *Int. J. Hydrogen Energy*, 2021, **46**, 8444–8453.
- 175 T. Bosserez, J. Rongé, J. Van Humbeeck, S. Haussener and J. Martens, *Oil Gas Sci. Technol.*, 2015, **70**, 877–889.
- 176 E. Kemppainen, A. Bodin, B. Sebok, T. Pedersen, B. Seger, B. Mei, D. Bae, P. C. K. Vesborg, J. Halme and O. Hansen, *Energy Environ. Sci.*, 2015, **8**, 2991–2999.
- 177 Y. Liu, L. Ren, Z. Zhang, X. Qi, H. Li and J. Zhong, *Sci. Rep.*, 2016, **6**, 22516.
- 178 Y. Liu, L. Ren, Z. Zhang, X. Qi, H. Li and J. Zhong, *Sci. Rep.*, 2016, **6**, 22516.
- 179 D. Siegmund, S. Metz, V. Peinecke, T. E. Warner, C. Cremers, A. Grevé, T. Smolinka, D. Segets and U.-P. Apfel, *JACS Au*, 2021, **1**, 527–535.
- 180 J. Wang, H.-x. Zhong, Z.-l. Wang, F.-l. Meng and X.-b. Zhang, *ACS Nano*, 2016, **10**, 2342–2348.
- 181 B. Konkana, K. Junge Puring, I. Sinev, S. Piontek, O. Khavryuchenko, J. P. Dürholt, R. Schmid, H. Tüysüz, M. Muhler and W. Schuhmann, *Nat. Commun.*, 2016, **7**, 1–8.
- 182 D. Siegmund, N. Blanc, M. Smialkowski, K. Tschulik and U. P. Apfel, *ChemElectroChem*, 2020, **7**, 1514–1527.
- 183 A. Grimm, W. A. de Jong and G. J. Kramer, *Int. J. Hydrogen Energy*, 2020, **45**, 22545–22555.
- 184 K. Zhang, M. B. McDonald, I. E. Genina and P. T. Hammond, *Chem. Mater.*, 2018, **30**, 6420–6430.
- 185 S. Zhang, C. Yin, D. Xing, D. Yang and X. Jian, *J. Membr. Sci.*, 2010, **363**, 243–249.
- 186 D. Chen, M. A. Hickner, E. Agar and E. C. Kumbur, *Electrochem. Commun.*, 2013, **26**, 37–40.
- 187 A. Berger, R. Segalman and J. Newman, *Energy Environ. Sci.*, 2014, **7**, 1468–1476.
- 188 J. R. McKone, N. S. Lewis and H. B. Gray, *Chem. Mater.*, 2013, **26**, 407–414.
- 189 N. M. Vargas-Barbosa, G. M. Geise, M. A. Hickner and T. E. Mallouk, *ChemSusChem*, 2014, **7**, 3017–3020.
- 190 D. A. Vermaas, M. Sassenburg and W. A. Smith, *J. Mater. Chem. A*, 2015, **3**, 19556–19562.
- 191 T. Lopes, L. Andrade, H. A. Ribeiro and A. Mendes, *Int. J. Hydrogen Energy*, 2010, **35**, 11601–11608.
- 192 A. Landman, H. Dotan, G. E. Shter, M. Wullenkord, A. Houaijia, A. Maljusch, G. S. Grader and A. Rothschild, *Nat. Mater.*, 2017, **16**, 646–651.
- 193 J. W. Ager Iii, M. Shaner, K. Walczak, I. D. Sharp and S. Ardo, *Energy Environ. Sci.*, 2015, **8**, 2811–2824.
- 194 M. A. Modestino and S. Haussener, *Annu. Rev. Chem. Biomol. Eng.*, 2015, **6**, 13–34.
- 195 T. J. Jacobsson, V. Fjallstrom, M. Edoff and T. Edvinsson, *Energy Environ. Sci.*, 2014, **7**, 2056–2070.
- 196 S. Chu, W. Li, Y. Yan, T. Hamann, I. Shih, D. Wang and Z. Mi, *Nano Futures*, 2017, **1**, 022001.
- 197 M.-K. Son, *Energies*, 2021, **14**, 7422.
- 198 Y. Chen, Y. Liu, F. Wang, X. Guan and L. Guo, *J. Energy Chem.*, 2021, **61**, 469–488.
- 199 K. R. Tolod, S. Hernández and N. Russo, *Catalysts*, 2017, **7**, 13.
- 200 Y. Tachibana, L. Vayssieres and J. R. Durrant, *Nat. Photonics*, 2012, **6**, 511–518.
- 201 A. Hankin, F. Bedoya-Lora, C. Ong, J. Alexander, F. Petter and G. Kelsall, *Energy Environ. Sci.*, 2017, **10**, 346–360.



- 202 T. Lopes, L. Andrade, F. Le Formal, M. Gratzel, K. Sivula and A. Mendes, *Phys. Chem. Chem. Phys.*, 2014, **16**, 16515–16523.
- 203 L. Han, F. F. Abdi, R. van de Krol, R. Liu, Z. Huang, H. J. Lewerenz, B. Dam, M. Zeman and A. H. Smets, *ChemSusChem*, 2014, **7**, 2832–2838.
- 204 A. I. Pereira, J. Martins, C. J. Tavares, L. Andrade and A. Mendes, *Appl. Surf. Sci.*, 2017, **423**, 549–556.
- 205 S.-Y. Park, M.-g Kim, J. Jung, J. Heo, E. M. Hong, S. M. Choi, J.-Y. Lee, S. Cho, K. Hong and D. C. Lim, *J. Power Sources*, 2017, **341**, 411–418.
- 206 D. Ivanou, J. Capitão, J. Maçaira, A. I. Pereira and A. Mendes, *J. Electrochem. Soc.*, 2018, **165**, H1040.
- 207 A. Vilanova, T. Lopes and A. Mendes, *J. Power Sources*, 2018, **398**, 224–232.
- 208 X. Yao, D. Wang, X. Zhao, S. Ma, P. S. Bassi, G. Yang, W. Chen, Z. Chen and T. Sritharan, *Energy Technol.*, 2018, **6**, 100–109.
- 209 S. Dilger, M. Trottmann and S. Pokrant, *ChemSusChem*, 2019, **12**, 1931–1938.
- 210 M. Huang, W. Lei, M. Wang, S. Zhao, C. Li, M. Wang and H. Zhu, *J. Mater. Chem. A*, 2020, **8**, 3845–3850.
- 211 A. Landman, R. Halabi, P. Dias, H. Dotan, A. Mehlmann, G. E. Shter, M. Halabi, O. Naseraldeem, A. Mendes and G. S. Grader, *Joule*, 2020, **4**, 448–471.
- 212 C. Acar and I. Dincer, *Int. J. Hydrogen Energy*, 2016, **41**, 7950–7959.
- 213 A. Eftekhari, V. J. Babu and S. Ramakrishna, *Int. J. Hydrogen Energy*, 2017, **42**, 11078–11109.
- 214 D. Hansora, D. Cherian, R. Mehrotra, J.-W. Jang and J. S. Lee, *Joule*, 2023, **7**, 884–919.
- 215 C. Carver, Z. Ulissi, C. Ong, S. Dennison, G. Kelsall and K. Hellgardt, *Int. J. Hydrogen Energy*, 2012, **37**, 2911–2923.
- 216 C. K. Ong, PhD thesis, Imperial College, London, 2013.
- 217 A. Vilanova, S. Miranda, A. Apolinário, L. Andrade, T. Lopes and A. Mendes, Public report on PEC device design University of Porto – Engineering Faculty, 2016.
- 218 G. Saracco, Report on the study on the potential future penetration of the ArtipHyction technology in the devised or in spin-off fields, Instituto Politécnico de Turim, Artiphyction: Fully artificial photo-electrochemical device for low temperature hydrogen production, 2015.
- 219 I. Holmes-Gentle, S. Tembhurne, C. Suter and S. Haussener, *Nat. Energy*, 2023, 1–11.
- 220 T. Lopes, *Characterization and Phenomenological Modeling of Photoelectrochemical Cells for Hydrogen Production from Solar Energy*, PhD thesis, UPORTO-FEUP, Porto, Portugal, 2014, <https://hdl.handle.net/10216/99846>.
- 221 J. Ihssen, A. Braun, G. Faccio, K. Gajda-Schranz and L. Thony-Meyer, *Curr. Protein Pept. Sci.*, 2014, **15**, 374–384.
- 222 A. Braun, M. Diale, K. Maabong and R. Toth, Safe And Decentralised Hydrogen Fuel Production And Storage For Residential Building And Mobility Applications, 2016.
- 223 A. Braun, M. M. Diale, K. D. Maabong and R. Toth, *Safe and decentralised solar hydrogen fuel production and storage for residential building and mobility applications*, 6th Int'l Disaster Risk Conf., 2016, pp. 102–105.
- 224 K. Walczak, Y. Chen, C. Karp, J. W. Beeman, M. Shaner, J. Spurgeon, I. D. Sharp, X. Amashukeli, W. West and J. Jin, *ChemSusChem*, 2015, **8**, 544–551.
- 225 N. S. Lewis, *Nat. Nanotechnol.*, 2016, **11**, 1010–1019.
- 226 J. M. Spurgeon, M. G. Walter, J. Zhou, P. A. Kohl and N. S. Lewis, *Energy Environ. Sci.*, 2011, **4**, 1772–1780.
- 227 A. Rothschild, G. Grader, G. Shter, A. Landman and H. Dotan, Methods and system for hydrogen production by water electrolysis, *US Pat.*, US10487408B2, 2015.
- 228 S. Miranda, A. Vilanova, T. Lopes and A. Mendes, *RSC Adv.*, 2017, **7**, 29665–29671.
- 229 C. Moon and B. Shin, *Discov. Mater.*, 2022, **2**, 5.
- 230 I. Y. Ahmet, S. Berglund, A. Chemseddine, P. Bogdanoff, R. F. Präg, F. F. Abdi and R. van de Krol, *Adv. Energy Sustainability Res.*, 2020, **1**, 2000037.
- 231 J. Zhu, J. B. Guðmundsdóttir, R. Strandbakke, K. G. Both, T. Aarholt, P. A. Carvalho, M. H. Sørby, I. J. Jensen, M. N. Guzik and T. Norby, *ACS Appl. Mater. Interfaces*, 2021, **13**, 20313–20325.
- 232 R. Sathre, C. D. Scown, W. R. Morrow, J. C. Stevens, I. D. Sharp, J. W. Ager, K. Walczak, F. A. Houle and J. B. Greenblatt, *Energy Environ. Sci.*, 2014, **7**, 3264–3278.
- 233 T. A. Kistler, M. Y. Um and P. Agbo, *J. Electrochem. Soc.*, 2020, **167**, 066502.
- 234 M. Wullenkord, C. Spenke, A. Vilanova, T. Lopes and A. Mendes, PECDEMO Project Deliverable Report D6.4: Public Report on Performance of the Large Area Prototype Array, 2017. Available online: [https://archiveweb.epfl.ch/pecdemo.epfl.ch/files/content/sites/pecdemo/files/public%20files/PECDEMO%20Deliverable%206.4\\_DLR\\_UPorto.pdf](https://archiveweb.epfl.ch/pecdemo.epfl.ch/files/content/sites/pecdemo/files/public%20files/PECDEMO%20Deliverable%206.4_DLR_UPorto.pdf) (accessed on 10 August 2023).
- 235 J. L. Young, M. A. Steiner, H. Döscher, R. M. France, J. A. Turner and T. G. Deutsch, *Nat. Energy*, 2017, **2**, 17028.
- 236 E. L. Miller, R. E. Rocheleau and X. M. Deng, *Int. J. Hydrogen Energy*, 2003, **28**, 615–623.
- 237 C. Xiang, A. Z. Weber, S. Ardo, A. Berger, Y. Chen, R. Coridan, K. T. Fountaine, S. Haussener, S. Hu and R. Liu, *Angew. Chem., Int. Ed.*, 2016, **55**, 12974.
- 238 C. Xiang, A. Z. Weber, S. Ardo, A. Berger, Y. Chen, R. Coridan, K. T. Fountaine, S. Haussener, S. Hu, R. Liu, N. S. Lewis, M. A. Modestino, M. M. Shaner, M. R. Singh, J. C. Stevens, K. Sun and K. Walczak, *Angew. Chem., Int. Ed.*, 2016, **55**, 12974–12988.
- 239 X. Deng and L. Xu, *Integrated photoelectrochemical cell and system having a liquid electrolyte*, WO2004050961A1, University of Toledo, Ohio, 2005.
- 240 X. Deng and L. Xu, *Interconnected photoelectrochemical cell*, US20080223439A1, Midwest Optoelectronics LLC and University of Toledo, Ohio, 2008.
- 241 M. Grätzel and J. Augustynski, *Tandem Cell for Water Cleavage by Visible Light*, WO0102624 (A1), EPFL, Switzerland, 2001.
- 242 L. Pan, Y. Liu, L. Yao, D. Ren, K. Sivula, M. Grätzel and A. Hagfeldt, *Nat. Commun.*, 2020, **11**, 318.
- 243 M. Cregut, M. Bedas, M.-J. Durand and G. Thouand, *Biotechnol. Adv.*, 2013, **31**, 1634–1647.





- 244 Z. Huaiwei and H. Xin, *Resour., Conserv. Recycl.*, 2011, **55**, 745–754.
- 245 M. Y. Khalid, Z. U. Arif, W. Ahmed and H. Arshad, *Sustainable Mater. Technol.*, 2022, **31**, e00382.
- 246 G. Gahleitner, *Int. J. Hydrogen Energy*, 2013, **38**, 2039–2061.
- 247 R. Boudries, *Int. J. Hydrogen Energy*, 2013, **38**, 11507–11518.
- 248 A. Hussain, S. M. Arif and M. Aslam, *Renewable Sustainable Energy Rev.*, 2017, **71**, 12–28.
- 249 M. S. Guney, *Renewable Sustainable Energy Rev.*, 2016, **57**, 776–785.
- 250 D. Azzouzi, B. Boumeddane and A. Abene, *Renewable Energy*, 2017, **106**, 111–121.
- 251 A. Kumar, O. Prakash and A. Dube, *Renewable Sustainable Energy Rev.*, 2017, **79**, 304–307.
- 252 A. Algarue, S. Mahmoud and R. Al-Dadah, *Energy Procedia*, 2014, **61**, 2375–2378.
- 253 S. Caron, M. Röger and M. Wullenkord, *Energies*, 2020, **13**, 5196.
- 254 S. Tembhurne and S. Haussener, *J. Electrochem. Soc.*, 2016, **163**, H999–H1007.
- 255 R. Abbas, M. Montes, A. Rovira and J. Martínez-Val, *Sol. Energy*, 2016, **124**, 198–215.
- 256 V. Jebasingh and G. J. Herbert, *Renewable Sustainable Energy Rev.*, 2016, **54**, 1085–1091.
- 257 H. Beltagy, D. Semmar, C. Lehaut and N. Said, *Renewable Energy*, 2017, **101**, 782–793.
- 258 G. Zhu, T. Wendelin, M. J. Wagner and C. Kutscher, *Sol. Energy*, 2014, **103**, 639–652.
- 259 Q. Xuan, G. Li, G. Pei, J. Ji, Y. Su and B. Zhao, *Sol. Energy*, 2017, **158**, 808–818.
- 260 E. Bellos, *Therm. Sci. Eng. Prog.*, 2019, **10**, 112–137.
- 261 M. Wullenkord, C. Jung and C. Sattler, 2014, **45868**, V001T02A034.
- 262 M. Wullenkord and C. Jung, 2015, **ISF-1**, E: Po-4.
- 263 J. P. Bijarniya, K. Sudhakar and P. Baredar, *Renewable Sustainable Energy Rev.*, 2016, **63**, 593–603.
- 264 F. Dähler, M. Wild, R. Schäppi, P. Haueter, T. Cooper, P. Good, C. Larrea, M. Schmitz, P. Furler and A. Steinfeld, *Sol. Energy*, 2018, **170**, 568–575.
- 265 G. Muthu, S. Shanmugam and A. Veerappan, *Energy Procedia*, 2014, **54**, 2–10.
- 266 L. Xiao, F.-W. Guo, S.-Y. Wu and Z.-L. Chen, *Renewable Energy*, 2020, **145**, 878–892.
- 267 D. Esteve, C. Ganibal, D. Steinmetz and A. Vialaron, *Int. J. Hydrogen Energy*, 1982, **7**, 711–716.
- 268 A. Fallisch, L. Schellhase, J. Fresko, M. Zechmeister, M. Zedda, J. Ohlmann, L. Zielke, N. Paust and T. Smolinka, *Int. J. Hydrogen Energy*, 2017, **42**, 13544–13553.
- 269 G. Peharz, F. Dimroth and U. Wittstadt, *Int. J. Hydrogen Energy*, 2007, **32**, 3248–3252.
- 270 P. Moraitis, R. Schropp and W. Van Sark, *Opt. Mater.*, 2018, **84**, 636–645.
- 271 W. G. Van Sark, K. W. Barnham, L. H. Slooff, A. J. Chatten, A. Büchtemann, A. Meyer, S. J. McCormack, R. Koole, D. J. Farrell and R. Bose, *Opt. Express*, 2008, **16**, 21773–21792.
- 272 M. G. Debije and P. P. Verbunt, *Adv. Energy Mater.*, 2012, **2**, 12–35.
- 273 M. D. Hughes, C. Maher, D.-A. Borca-Tasciuc, D. Polanco and D. Kaminski, *Renewable Energy*, 2013, **52**, 266–272.
- 274 G. Panzeri, E. Tatsi, G. Griffini and L. Magagnin, *ACS Appl. Energy Mater.*, 2020, **3**, 1665–1671.
- 275 G. Liu, B. Sun, H. Li, Y. Wang and H. Zhao, *J. Mater. Chem. A*, 2019, **7**, 18529–18537.
- 276 M. Dumortier and S. Haussener, *Energy Environ. Sci.*, 2015, **8**, 3069–3082.
- 277 J. Jia, L. C. Seitz, J. D. Benck, Y. Huo, Y. Chen, J. W. D. Ng, T. Bilir, J. S. Harris and T. F. Jaramillo, *Nat. Commun.*, 2016, **7**, 13237.
- 278 J. S. Price, X. Sheng, B. M. Meulblok, J. A. Rogers and N. C. Giebink, *Nat. Commun.*, 2015, **6**, 1–8.
- 279 C. A. Rodriguez, M. A. Modestino, D. Psaltis and C. Moser, *Energy Environ. Sci.*, 2014, **7**, 3828–3835.
- 280 G. Segev, H. Dotan, K. D. Malviya, A. Kay, M. T. Mayer, M. Grätzel and A. Rothschild, *Adv. Energy Mater.*, 2016, **6**, 1500817.
- 281 M. A. Khan, I. Al-Shankiti, A. Ziani, N. Wehbe and H. Idriss, *Angew. Chem.*, 2020, **132**, 14912–14918.
- 282 A. Mojiri, R. Taylor, E. Thomsen and G. Rosengarten, *Renewable Sustainable Energy Rev.*, 2013, **28**, 654–663.
- 283 M. A. Green and A. Ho-Baillie, *Prog. Photovoltaics Res. Appl.*, 2010, **18**, 42–47.
- 284 Y. Bicer, A. F. V. Sprotte and I. Dincer, *Appl. Energy*, 2017, **197**, 169–182.
- 285 S. Tembhurne and S. Haussener, *Sustainable Energy Fuels*, 2019, **3**, 1297–1306.
- 286 H. Zhang, L. V. Besteiro, J. Liu, C. Wang, G. S. Selopal, Z. Chen, D. Barba, Z. M. Wang, H. Zhao and G. P. Lopinski, *Nano Energy*, 2020, 105416.
- 287 A. Leblebici, P. Mayor, M. Rajman and G. De Micheli, *Nano-Tera. ch*, Springer, 2019, pp.109–137.
- 288 V. Zagolla, D. Dominé, E. Tremblay and C. Moser, *Opt. Express*, 2014, **22**, A1880–A1894.
- 289 V. Zagolla, C. Moser and J.-E. Moser, Self-tracking solar concentrator device, US20150063751A1, 2017.
- 290 Q. Wang, C. Pornrungrroj, S. Linley and E. Reisner, *Nat. Energy*, 2022, **7**, 13–24.
- 291 M. E. Oruc, A. V. Desai, P. J. Kenis and R. G. Nuzzo, *Appl. Energy*, 2016, **164**, 294–302.
- 292 S. Rashidi, N. Karimi, B. Sunden, K. C. Kim, A. G. Olabi and O. Mahian, *Prog. Energy Combust. Sci.*, 2022, **88**, 100966.
- 293 C. Pornrungrroj, V. Andrei and E. Reisner, *J. Am. Chem. Soc.*, 2023, **145**(25), 13709–13714.
- 294 R. Boudries, A. Khellaf, A. Aliane, L. Ihaddaden and F. Khida, *Int. J. Hydrogen Energy*, 2014, **39**, 15188–15195.
- 295 H. G. Cha and K.-S. Choi, *Nat. Chem.*, 2015, **7**, 328–333.
- 296 X. Lu, S. Xie, H. Yang, Y. Tong and H. Ji, *Chem. Soc. Rev.*, 2014, **43**, 7581–7593.
- 297 H. Luo, J. Barrio, N. Sunny, A. Li, L. Steier, N. Shah, I. E. Stephens and M. M. Titirici, *Adv. Energy Mater.*, 2021, **11**, 2101180.
- 298 N. Ibrahim, S. K. Kamarudin and L. Minggu, *J. Power Sources*, 2014, **259**, 33–42.





- 299 J. Plácido and S. Capareda, *Bioresour Bioprocess*, 2016, **3**, 1–12.
- 300 X. Han, H. Sheng, C. Yu, T. W. Walker, G. W. Huber, J. Qiu and S. Jin, *ACS Catal.*, 2020, **10**, 6741–6752.
- 301 N. Perini, C. Hessel, J. L. Bott-Neto, C. T. Pires, P. S. Fernandez and E. Sitta, *J. Solid State Electrochem.*, 2021, **25**, 1101–1110.
- 302 Y. Wang, G. Bu, X. Geng, Z. Zhu, P. Cui and Z. Liao, *J. Cleaner Prod.*, 2019, **218**, 212–224.
- 303 G. Ramis, E. Bahadori and I. Rossetti, *Int. J. Hydrogen Energy*, 2021, **46**, 12105–12116.
- 304 C.-W. Huang, B.-S. Nguyen, J. C.-S. Wu and V.-H. Nguyen, *Int. J. Hydrogen Energy*, 2020, **45**, 18144–18159.
- 305 D. Antón-García, E. Edwardes Moore, M. A. Bajada, A. Eisenschmidt, A. R. Oliveira, I. A. Pereira, J. Warnan and E. Reisner, *Nat. Synth.*, 2022, **1**, 77–86.
- 306 C. Pulignani, C. A. Mesa, S. A. Hillman, T. Uekert, S. Giménez, J. R. Durrant and E. Reisner, *Angew. Chem., Int. Ed.*, 2022, **61**, e202211587.
- 307 G. G. N. Thushari and J. D. M. Senevirathna, *Heliyon*, 2020, **6**, e04709.
- 308 M. Demertzi, R. P. Silva, B. Neto, A. C. Dias and L. Arroja, *J. Cleaner Prod.*, 2016, **112**, 1985–1994.
- 309 R. Sathre, J. B. Greenblatt, K. Walczak, I. D. Sharp, J. C. Stevens, J. W. Ager and F. A. Houle, *Energy Environ. Sci.*, 2016, **9**, 803–819.
- 310 P. Nikolaidis and A. Poullikkas, *Renewable Sustainable Energy Rev.*, 2017, **67**, 597–611.
- 311 M. Yu, K. Wang and H. Vredenburg, *Int. J. Hydrogen Energy*, 2021, **46**, 21261–21273.
- 312 D. Girelen, E. Taibi and R. Miranda, Hydrogen: a renewable energy perspective Report 978-92-9260-151-5, International Renewable Energy Agency, 2019.
- 313 I. Dincer, *Int. J. Hydrogen Energy*, 2012, **37**, 1954–1971.
- 314 J. M. M. Arcos and D. M. Santos, *Gases*, 2023, **3**, 25–46.
- 315 Y. Lovcha, A. Perez-Laborda and I. Sikora, *Appl. Energy*, 2022, **305**, 117903.
- 316 M. Sato, R. Rafaty, R. Calel and M. Grubb, *Wiley Interdiscip. Rev. Clim. Change*, 2022, **13**, e796.
- 317 A. Ajanovic, M. Sayer and R. Haas, *Int. J. Hydrogen Energy*, 2022, **47**, 24136–24154.
- 318 F. Pruvost, S. Cloete, C. A. del Pozo and A. Zaabout, *Energy Convers. Manage.*, 2022, **274**, 116458.
- 319 J. F. George, V. P. Müller, J. Winkler and M. Ragwitz, *Energy Policy*, 2022, **167**, 113072.
- 320 A. Midilli, H. Kucuk, M. E. Topal, U. Akbulut and I. Dincer, *Int. J. Hydrogen Energy*, 2021, **46**, 25385–25412.
- 321 A. Iulianelli, S. Liguori, J. Wilcox and A. Basile, *Catal. Rev.*, 2016, **58**, 1–35.
- 322 I. Dincer and C. Zamfirescu, *Int. J. Hydrogen Energy*, 2012, **37**, 16266–16286.
- 323 L. Cao, K. Iris, X. Xiong, D. C. Tsang, S. Zhang, J. H. Clark, C. Hu, Y. H. Ng, J. Shang and Y. S. Ok, *Environ. Res.*, 2020, **186**, 109547.
- 324 O. Bereketidou and M. Goula, *Catal. Today*, 2012, **195**, 93–100.
- 325 L. Alves, V. Pereira, T. Lagarteira and A. Mendes, *Renewable Sustainable Energy Rev.*, 2021, **137**, 110465.
- 326 M. R. Giraldi, J.-L. François and C. Martin-del-Campo, *Int. J. Hydrogen Energy*, 2015, **40**, 4019–4033.
- 327 O. Schmidt, A. Gambhir, I. Staffell, A. Hawkes, J. Nelson and S. Few, *Int. J. Hydrogen Energy*, 2017, **42**, 30470–30492.
- 328 T. L. Gibson and N. A. Kelly, *Int. J. Hydrogen Energy*, 2008, **33**, 5931–5940.
- 329 B. Olateju, A. Kumar and M. Secanell, *Int. J. Hydrogen Energy*, 2016, **41**, 8755–8776.
- 330 C. Acar and I. Dincer, *Int. J. Hydrogen Energy*, 2022, **47**, 40118–40137.
- 331 D. Burchart, M. Gazda-Grzywacz, P. Grzywacz, P. Burmistrz and K. Zarębska, *Energies*, 2022, **16**, 383.
- 332 J. Li and W. Cheng, *Int. J. Hydrogen Energy*, 2020, **45**, 27979–27993.
- 333 A. Verma and A. Kumar, *Appl. Energy*, 2015, **147**, 556–568.
- 334 H. Liu, W. Guo and S. Liu, *Energy*, 2022, **258**, 125001.
- 335 Y. Ju and C.-H. Lee, *Energy Convers. Manage.*, 2017, **143**, 123–136.
- 336 K. Bareiß, C. de la Rua, M. Möckl and T. Hamacher, *Appl. Energy*, 2019, **237**, 862–872.
- 337 C. Acar and I. Dincer, *Int. J. Energy Res.*, 2015, **39**(13), 1757–1768.
- 338 G. Di Marcoberardino, X. Liao, A. Dauriat, M. Binotti and G. Manzolini, *Processes*, 2019, **7**, 86.
- 339 A. Mehmeti, A. Angelis-Dimakis, G. Arampatzis, S. J. McPhail and S. Ulgiati, *Environments*, 2018, **5**, 24.
- 340 N. Hajjaji, S. Martinez, E. Trably, J.-P. Steyer and A. Helias, *Int. J. Hydrogen Energy*, 2016, **41**, 6064–6075.
- 341 R. Bhandari, C. A. Trudewind and P. Zapp, *J. Cleaner Prod.*, 2014, **85**, 151–163.
- 342 M. Ji and J. Wang, *Int. J. Hydrogen Energy*, 2021, **46**, 38612–38635.
- 343 A. E. Karaca and I. Dincer, *Chemosphere*, 2023, **337**, 139367.
- 344 C. Yan, A. Yang, I.-L. Chien, W. Shen and J. Ren, *Energy Convers. Manage.*, 2019, **199**, 111963.
- 345 A. Valente, D. Iribarren, J.-L. Gálvez-Martos and J. Dufour, *Sci. Total Environ.*, 2019, **650**, 1465–1475.
- 346 B. Lee, H. Chae, N. H. Choi, C. Moon, S. Moon and H. Lim, *Int. J. Hydrogen Energy*, 2017, **42**, 6462–6471.
- 347 A. Maljush, M. Wullenkord, E. Smirnova and C. Jung, Report on the results of life-cycle analysis, economic analysis and benchmarking, PECDEMO, 2017.
- 348 N. Hajjaji, M.-N. Pons, A. Houas and V. Renaudin, *Energy Policy*, 2012, **42**, 392–399.
- 349 A. Maljusch and M. Wullenkord, *Advances in Photoelectrochemical Water Splitting*, Royal Society of Chemistry, 2018, pp.266–284.
- 350 G. Li, S. Wang, J. Zhao, H. Qi, Z. Ma, P. Cui, Z. Zhu, J. Gao and Y. Wang, *Energy*, 2020, 117488.
- 351 I. Dincer and C. Acar, *Int. J. Hydrogen Energy*, 2015, **40**, 11094–11111.
- 352 R. S. El-Emam and H. Özcan, *J. Cleaner Prod.*, 2019, **220**, 593–609.
- 353 A. Valente, D. Iribarren and J. Dufour, *Int. J. Hydrogen Energy*, 2019, **44**, 19426–19433.



- 354 D. Iribarren, A. Susmozas, F. Petrakopoulou and J. Dufour, *J. Cleaner Prod.*, 2014, **69**, 165–175.
- 355 M. Kumar, A. O. Oyedun and A. Kumar, *Int. J. Hydrogen Energy*, 2019, **44**, 10384–10397.
- 356 H. R. Sara, B. Enrico, V. Mauro and N. Vincenzo, *Energy Procedia*, 2016, **101**, 806–813.
- 357 J. Yao, M. Kraussler, F. Benedikt and H. Hofbauer, *Energy Convers. Manage.*, 2017, **145**, 278–292.
- 358 E. Rillo, M. Gandiglio, A. Lanzini, S. Bobba, M. Santarelli and G. Blengini, *Energy*, 2017, **126**, 585–602.
- 359 A. Patyk, T. M. Bachmann and A. Brisse, *Int. J. Hydrogen Energy*, 2013, **38**, 3865–3880.
- 360 L. B. Braga, J. L. Silveira, M. E. Da Silva, C. E. Tuna, E. B. Machin and D. T. Pedroso, *Renewable Sustainable Energy Rev.*, 2013, **28**, 166–173.
- 361 Y. M. Camacho, S. Bensaid, G. Piras, M. Antonini and D. Fino, *Clean Technol. Environ. Policy*, 2017, **19**, 1437–1447.
- 362 M. Katebah and P. Linke, *Clean. Eng. Technol.*, 2022, **10**, 100552.
- 363 H. Song, Y. Liu, H. Bian, M. Shen and X. Lin, *Energy Convers. Manage.*, 2022, **258**, 115513.
- 364 A. Al-Qahtani, B. Parkinson, K. Hellgardt, N. Shah and G. Guillen-Gosalbez, *Appl. Energy*, 2021, **281**, 115958.
- 365 I. J. Okeke, B. A. Saville and H. L. MacLean, *Int. J. Hydrogen Energy*, 2023, **48**, 12581–12599.
- 366 A. Oni, K. Anaya, T. Giwa, G. Di Lullo and A. Kumar, *Energy Convers. Manage.*, 2022, **254**, 115245.
- 367 F. Kersch, A. Stary, S. Gleis, A. Ulrich, H. Klein and H. Spliethoff, *Int. J. Hydrogen Energy*, 2021, **46**, 19897–19912.
- 368 S. Timmerberg, M. Kaltschmitt and M. Finkbeiner, *Energy Convers. Manage.: X*, 2020, **7**, 100043.
- 369 N. Sánchez-Bastardo, R. Schlögl and H. Ruland, *Ind. Eng. Chem. Res.*, 2021, **60**, 11855–11881.
- 370 A. E. Karaca, I. Dincer and J. Gu, *Int. J. Hydrogen Energy*, 2020, **45**, 22148–22159.
- 371 M. Sorrentino, A. Adamo and G. Nappi, *Energies*, 2019, **12**, 3224.
- 372 R. Anghilante, D. Colomar, A. Brisse and M. Marrony, *Int. J. Hydrogen Energy*, 2018, **43**, 20309–20322.
- 373 M. Lin and S. Haussener, *Sol. Energy*, 2017, **155**, 1389–1402.
- 374 O. Posdziech, K. Schwarze and J. Brabandt, *Int. J. Hydrogen Energy*, 2019, **44**, 19089–19101.
- 375 E. Giglio, A. Lanzini, M. Santarelli and P. Leone, *J. Energy Storage*, 2015, **1**, 22–37.
- 376 J. C. Koj, C. Wulf, A. Schreiber and P. Zapp, *Energies*, 2017, **10**, 860.
- 377 A. Valente, D. Iribarren and J. Dufour, *Sci. Total Environ.*, 2020, **728**, 138212.
- 378 N. Gerloff, *ACS Sustainable Chem. Eng.*, 2021, **9**, 10123–10141.
- 379 B. Lee, H.-S. Cho, H. Kim, D. Lim, W. Cho, C.-H. Kim and H. Lim, *J. Environ. Chem. Eng.*, 2021, **9**, 106349.
- 380 J. E. Mason and K. Zweibel, *Int. J. Hydrogen Energy*, 2007, **32**, 2743–2763.
- 381 E. Cetinkaya, I. Dincer and G. F. Naterer, *Int. J. Hydrogen Energy*, 2012, **37**, 2071–2080.
- 382 M. Fereidooni, A. Mostafaeipour, V. Kalantar and H. Goudarzi, *Renewable Sustainable Energy Rev.*, 2018, **82**, 415–423.
- 383 M. Reuß, J. Reul, T. Grube, M. Langemann, S. Calnan, M. Robinius, R. Schlattmann, U. Rau and D. Stolten, *Sustainable Energy Fuels*, 2019, **3**, 801–813.
- 384 S. Ghandehariun and A. Kumar, *Int. J. Hydrogen Energy*, 2016, **41**, 9696–9704.
- 385 T. Ayodele and J. Munda, *Int. J. Hydrogen Energy*, 2019, **44**, 17669–17687.
- 386 R. Fang, *Int. J. Hydrogen Energy*, 2019, **44**, 29399–29408.
- 387 M. Aguado, E. Ayerbe, C. Azcárate, R. Blanco, R. Garde, F. Mallor and D. M. Rivas, *Int. J. Hydrogen Energy*, 2009, **34**, 2845–2854.
- 388 J. B. Greenblatt, *Integrated Solar Fuel Generators*, 2018, vol. 22, p.43.
- 389 L. Xu, X. Deng, A. Abken, X. Cao, W. Du, A. Vijn, W. Ingler, C. Chen, Q. Fan and R. Collins, *Critical Research for Cost-Effective Photoelectrochemical Production of Hydrogen*, Midwest Optoelectronics, LLC, Toledo, OH (United States), 2014.
- 390 C. Acar and I. Dincer, *J. Cleaner Prod.*, 2019, **218**, 835–849.
- 391 G. Palmer, A. Roberts, A. Hoadley, R. Dargaville and D. Honnery, *Energy Environ. Sci.*, 2021, **14**, 5113–5131.
- 392 A. P. Ravikumar, M. Bazilian and M. E. Webber, *Nat. Energy*, 2022, 1–3.
- 393 I. Kougiyas, N. Taylor, G. Kakoulaki and A. Jäger-Waldau, *Renewable Sustainable Energy Rev.*, 2021, **144**, 111017.
- 394 S. Sadeghi, S. Ghandehariun and M. A. Rosen, *Energy*, 2020, **208**, 118347.
- 395 J. Tian, L. Yu, R. Xue, S. Zhuang and Y. Shan, *Appl. Energy*, 2022, **307**, 118205.
- 396 A hydrogen strategy for a climate-neutral Europe, European Commission, 2020. Available online: [https://energy.ec.europa.eu/system/files/2020-07/hydrogen\\_strategy\\_0.pdf](https://energy.ec.europa.eu/system/files/2020-07/hydrogen_strategy_0.pdf) (accessed on 3 August 2023).
- 397 I. Dincer and C. Acar, *Int. J. Hydrogen Energy*, 2018, **43**, 8579–8599.
- 398 A. Gatto, *Energy Res. Soc. Sci.*, 2022, **89**, 102639.
- 399 T. J. Jacobsson, V. Fjällström, M. Edoff and T. Edvinsson, *Sol. Energy Mater. Sol. Cells*, 2015, **134**, 185–193.
- 400 T. J. Jacobsson, *Energy Environ. Sci.*, 2018, **11**, 1977–1979.
- 401 W. Li, J. Zheng, B. Hu, H.-C. Fu, M. Hu, A. Veyssal, Y. Zhao, J.-H. He, T. L. Liu and A. Ho-Baillie, *Nat. Mater.*, 2020, **19**, 1326–1331.

