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Ultrasonic reduction: an unconventional route to exsolute Ag from perovskite La(Ag)FeO_{3-δ} for enhanced catalytic oxidation activity†

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We explore an uncommon ultrasonic reduction method to exsolute Ag from perovskite La_{0.87}Ag_{0.03}FeO_{3-δ}, forming a composite with enhanced catalytic oxidation activity. Such a mild exsolution is based on the coupling effect of ultrasonic cavitation and reducible BH₄⁻, and holds great potential in the fields of energy and environment catalysis.

Perovskite oxide-based composites are widely considered to be key catalysts in the fields of energy and environment due to their strong electron transfer between components.^{1–10} Many efforts have been made to develop synthetic methods for perovskite oxide-based composites with tight combination, strong interaction, and uniform particle distribution.^{5–19} Despite several advances in this regard, harsh conditions such as high-pressure hydrogen atmosphere and high temperature are still the primary exsolution methods for forming composites. For example, under conditions of H₂ and 900 °C, Co³⁺ was exsolated from bulk La_{0.3}Sr_{0.7}Co_{0.07}Ti_{0.93}O_{3-δ} to the surface and formed nano-sized Co clusters that reduced the anode polarization resistance and improved the solid oxide fuel cell performance of La_{0.3}Sr_{0.7}TiO_{3-δ} as well.¹¹ Nano-alloy Fe–Ni can be exsolated from Sr₂Fe_{1.3}Ni_{0.2}Mo_{0.5}O_{6-δ} by H₂ annealing, showing enhanced oxygen evolution activity.¹⁴ Kim's team constructed an Ir/LaFe_{0.99}Ir_{0.01}O₃ composite by exsolution, and the composite showed good hydrogen production properties.¹⁷ It is noted that the classical thermal reduction exsolution method is conducted at a temperature above 500 °C and in H₂ atmosphere,^{11–13} and moreover, the pristine oxides tend to undergo structural distortion and even structure collapse during the exsolution process.²⁰ Nowadays, sustainable development requires environment-friendly exsolution methods without

involving harsh conditions such as high temperature and hydrogen atmosphere, which remains highly challenging.

Ultrasonic reduction exsolution could be promising for meeting such challenges, due to the coupling effects of ultrasonic cavitation^{21–23} and strengthened reducibility of NaBH₄ solution. High-intensity ultrasonic waves can generate a large number of tiny cavitation bubbles in the propagation medium,²⁴ while the collapse of the cavitation bubbles can create a local environment of high temperature, pressure, and shockwaves, providing a special physicochemical environment for chemical reactions. The ultrasonic waves strengthened the reducibility of the NaBH₄ solution, creating a reaction environment with local high temperature and pressure, and strong reducibility, which is similar to the classical high-pressure hydrogen atmosphere annealing exsolution. Even so, thermal reduction exsolution is carried out in a high temperature atmosphere, and thus the lattice oxygen could combine with active hydrogen to form water molecules and escape from the pristine oxide. In this case, a large number of cations and oxygen vacancies are generated in the reaction, which could lead to lattice collapse and even phase transformation.^{4,14,15} Fortunately, ultrasonic reduction exsolution is carried out in an aqueous solution at room temperature and pressure, and the solution environment prevents the escape of lattice oxygen after combining with active hydrogen, confirming the stability of the catalysts.^{25,26} Therefore, ultrasonic reduction exsolution could have potential for the fabrication of advanced composites.

In this work, we report an ultrasonic reduction exsolution that was successfully applied to the synthesis of composite Ag_{0.03-x}/La_{0.87}Ag_xH_yFeO_{3-δ} (LAFO-UR). Under ultrasonic waves, the active hydrogen was inserted into the lattice of perovskite LAFO, which then transfers electrons to Ag⁺ ions, leading to their reduction and exsolution from the lattice to form Ag nanoparticles on the surface of the perovskite, and construct a composite Ag_{0.03-x}/La_{0.87}Ag_xH_yFeO_{3-δ} (Fig. 1a). The crystal structures of the synthesized samples, including LaFeO₃ (LFO), LAFO, and LAFO-UR, were verified by X-ray diffraction (XRD)

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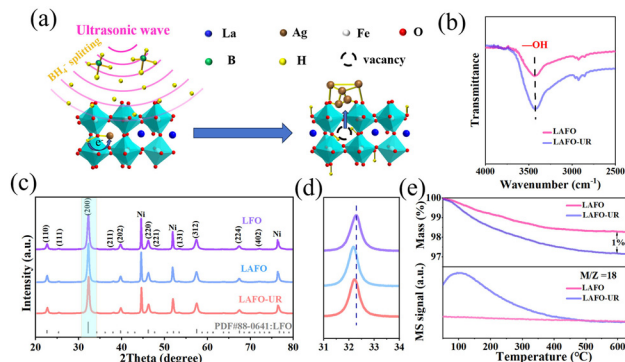


Fig. 1 (a) Schematic illustration of LAFO ultrasound reduction, (b) FT-IR data of LAFO and LAFO-UR, (c) XRD patterns of LFO, LAFO, and LAFO-UR (Ni is the internal standard), (d) enlarged XRD patterns in the interval of 31–34°, and (e) TG-MS data of LAFO and LAFO-UR.

patterns (Fig. 1b), which showed that all signal peaks of the synthesized LFO can be attributed to perovskite LaFeO_3 (PDF#88-0641), similar to other two samples LAFO and LAFO-UR. Even so, from the enlarged XRD patterns in the interval of 31°–34° (Fig. 1c), one can see that the peaks of LAFO and LAFO-UR shifted towards lower angles relative to pure perovskite LFO. This demonstrates the lattice expansion after Ag doping in the perovskite due to the larger radius of Ag^+ compared to La^{3+} . Additionally, LAFO-UR exhibited a smaller shift towards higher angles relative to LAFO, indicating that some of the doped Ag^+ ions were reduced and escaped from the perovskite lattice. The results of XRD refined data provide proof of the above conclusions (Fig. S3 and Table S1, ESI†).

The mechanism of ultrasonic reduction exsolution was investigated by Fourier Transform Infrared Spectroscopy (FT-IR) (Fig. 1e) and Thermogravimetric Mass Spectrometry (TG-MS) (Fig. 1d and Fig. S5, ESI†). The FT-IR analysis revealed that LAFO-UR contained more O–H species compared to LAFO, as evidenced by an observable characteristic peak at a wavelength of 3430 cm^{-1} (Fig. 1b). This conclusion is further supported by TG-MS data; the mass loss of LAFO-UR at 600 °C is 1% higher than that of LAFO (Fig. 1d), and this difference in mass loss is originated from the release of water from LAFO-UR, possibly formed by the condensation of O–H at high temperatures. The results suggest that hydrogen inserted could combine with lattice oxygen during the ultrasonic reduction process, leading to the formation of water, while lattice oxygen does not escape during the reaction process. The data of TG can indeed be used to calculate the amount of protons inserted into LAFO, and the molecular formula of LAFO-UR is $\text{Ag}_{0.03-x}\text{La}_{0.87}\text{Ag}_x\text{H}_{0.13}\text{FeO}_{3-\delta}$. Consequently, no oxygen vacancies were generated during the LAFO reduction process, as indicated by Electron Paramagnetic Resonance (EPR) (Fig. S6, ESI†).

The morphology and elemental distribution of the samples LFO, LAFO, and LAFO-UR were observed by Transmission Electron Microscope (TEM) images. Both LAFO and LAFO-UR exhibited particle sizes ranging from 15 to 50 nm, and no significant morphology change was observed after ultrasonic reduction. High Resolution Transmission Electron Microscope

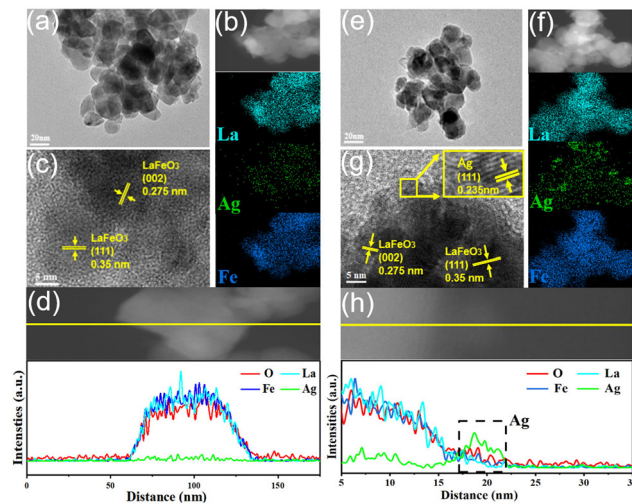


Fig. 2 (a) and (e) TEM images. (b) and (f) Elemental mapping. (c) and (g) HRTEM images. (d) and (h) Linear scan of LAFO and LAFO-UR.

(HRTEM) analysis revealed lattice spacings of 0.275 and 0.35 nm for LAFO and LAFO-UR, corresponding to the crystal planes (002) and (111) of perovskite LaFeO_3 , respectively. Additionally, small nanoparticles with a lattice spacing of 0.235 nm were found in LAFO-UR, which was consistent with the (111) crystal plane of Ag metal (Fig. 2g), suggesting the formation of Ag nanoparticles on the surface of LAFO. Elemental mapping illustrated that La and Fe were uniformly distributed in both LAFO and LAFO-UR (Fig. 2b and f); meanwhile, in LAFO-UR, parts of Ag (Fig. 2f) were significantly aggregated compared to LAFO (Fig. 2b), with the sizes of the Ag aggregations ranged from 3 to 5 nm, consistent with particle size of Ag nanoparticles observed in TEM images (Fig. S7, ESI†). This result is further confirmed by linear scanning. In LAFO, the changing trend of Ag content is in the same way as that of La and O (Fig. 2d). Nevertheless, in LAFO-UR (Fig. 2h and Fig. S8, ESI†), Ag shows an aggregated distribution, and the surface nanoparticles mainly consist of Ag element, which provides direct evidence for the exsolution of Ag.

The surface composition and structure of the samples were obtained by X-ray Photoelectron Spectroscopy (XPS). From Ag 3d spectra of LAFO and LAFO-UR in Fig. 3a, one can see a larger Ag^+ signal peak at 367.8 eV, in addition to the signal peak of Ag^0 at 368.5 eV for LAFO-UR, which indicates the presence of Ag nanoparticles after ultrasonic reduction exsolution in LAFO-UR. The O 1s spectrum (Fig. 3c) shows that the peak at 529.3 eV corresponds to Fe–O–Fe in the perovskite,²⁷ while the peak at 531.25 eV is attributed to O–H related species (Fe–O–H, H–O–H, etc.) on the surface of the perovskite. Obviously, the content of O–H on the surface of LAFO-UR is found to be significantly higher than that of LAFO. In addition, Fe 2p spectra indicate that the valence state of Fe is closer to +3 (Fig. 3b).

Synchrotron X-ray Absorption Fine Structure (XAFS) was used to study the electronic states of the samples LAFO and LAFO-UR. In Fig. 3d, for Ag K-edge spectra, the intensity of the white line peak is related to the valence state.²⁸ LAFO-UR



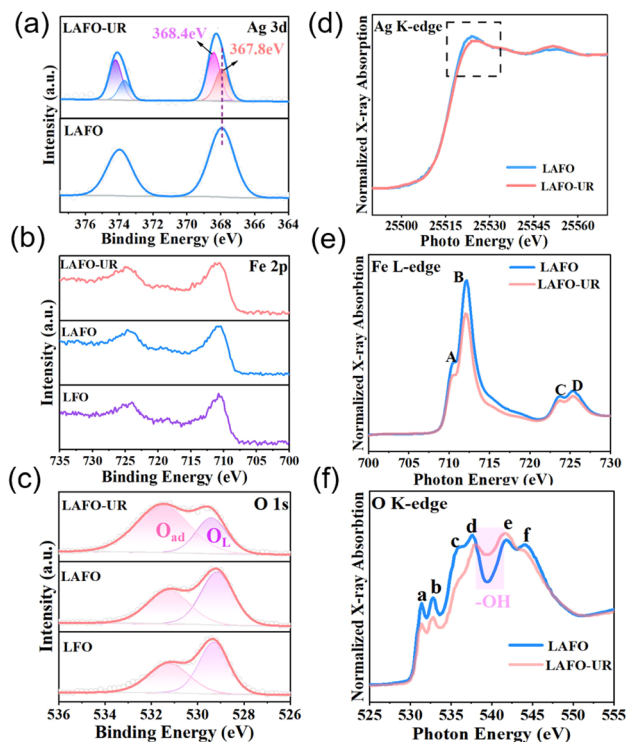


Fig. 3 (a) Ag 3d XPS spectra of LAFO and LAFO-UR, (b) Fe 2p and (c) O 1s XPS spectra of LAFO and LAFO-UR, relative to LFO; (d) Ag K-edge, (e) Fe L-edge, and (f) O K-edge XAFS spectra of LAFO and LAFO-UR.

exhibits a lower intensity of the white line peak in comparison with LAFO, indicating a significant decrease in valence state of Ag after ultrasonic reduction. Fig. 3e shows the normalized Fe L-edge spectra. Peaks A and C could be attributed to the Fe 3d–O 2p hybrid t_{2g} orbital, while peaks B and D could be assigned to the Fe 3d–O 2p hybrid e_g orbital.²⁹ The peak intensity of LAFO-UR is slightly lower than that of LAFO, indicating that the density of electronic states of Fe in LAFO-UR is higher than that of Fe in LAFO. Meanwhile, the ratio of peak B to peak A was calculated (Table S4, ESI†), and the ratio values of LAFO and LAFO-UR were 2.13 and 2.04, respectively. This indicates that the probability of electron transfer from the t_{2g} orbital to the e_g orbital increases in LAFO-UR, as reported elsewhere.²⁹ Therefore, the valence of Fe decreased after ultrasonic reduction. Fig. 3f is the normalized O K-edge spectra. Peaks a and b could be attributed to the orbitals t_{2g} and e_g in the Fe 3d–O 2p hybrid orbital. Peaks c and d could be associated with the La 5d–O 2p hybrid zone, while peaks e and f are derived from Fe 4sp and O 2p.³⁰ To further discuss the electronic states, we calculated the intensity ratio of peak b to peak a (Table S5, ESI†). The ratio values for LAFO and LAFO-UR are 1.07 and 1.11, respectively, and this result suggests a transfer of e_g orbital electrons in the Fe 3d–O 2p hybrid orbital to the t_{2g} orbital, leading to a decrease in the valence of O, like the observation reported elsewhere.²⁹ The main reason for the decrease of the O valence state is charge transfer from the Ag nanoparticles to the perovskite. The peak intensity of

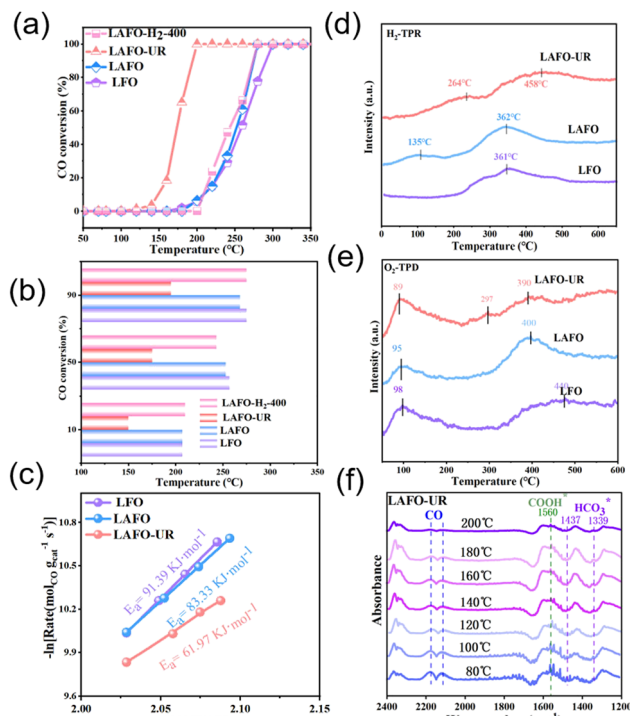


Fig. 4 (a) CO catalytic oxidation curves and (b) 2D histograms of 10%, 50%, and 90% CO conversion versus reaction temperature for the given samples; (c) Arrhenius plots, (d) H₂-TPR, and (e) O₂-TPD for samples LFO, LAFO, and LAFO-UR. (d) H₂-TPR and (e) O₂-TPD of LFO, LAFO, and LAFO-UR. (f) *In situ* DRIFTS spectra of LAFO-UR in a reactive atmosphere (1% CO, 20% O₂/He).

LAFO-UR at 538 ~ 542 eV (Fig. 3f) was enhanced because of the production of hydroxyl groups, as reported elsewhere.^{31,32}

The oxidation activity of the samples LFO, LAFO, LAFO-UR, and hydrogen-annealed LAFO (LAFO-H₂-400) was investigated, as shown in Fig. 4a. LAFO-UR exhibits the best catalytic performance among all the samples. Additionally, the 10%, 50%, and 90% conversion temperatures of LAFO-UR are 150 °C, 175 °C, and 195 °C, respectively (Fig. 4b), all of which are lower than those of LAFO and LAFO-H₂-400. A linear Arrhenius relationship between ln(rate) and the reciprocal of the temperature (1000/T) was obtained for LFO, LAFO and LAFO-UR (Fig. 4c). Activation energy indirectly reflects catalytic activity.³³ The activation energy of LAFO-UR is calculated to be 61.97 kJ mol⁻¹, which is lower than those of LFO and LAFO, indicating that the catalytic CO oxidation of LAFO-UR is easier to occur at low temperatures.

H₂-TPR and O₂-TPD were used to characterize the change in the chemical nature of the samples. In the H₂-TPR analysis (Fig. 4d), it is observed that the reduction temperature of Ag⁺ (135 °C) is below 300 °C, while the temperature range for reducing Fe³⁺ to Fe²⁺ is 300–500 °C. The reduction temperatures of Ag⁺ and Fe³⁺/Fe²⁺ in LAFO-UR have been significantly increased compared to LAFO. This result demonstrates that Ag⁺ and Fe³⁺/Fe²⁺ are activated during ultrasonic reduction, and are easily reduced at lower temperatures. O₂-TPD is further used to understand the oxygen species desorption behavior of the



samples (Fig. 4e). Low temperature desorption of physically adsorbed molecular O₂ and adsorbed O²⁻ (50–200 °C), and intermediate temperature desorption of O⁻, O²⁻ and surface lattice oxygen (300–500 °C) are shown in Fig. 4e. The low and medium temperature ranges of LAFO-UR shifted towards lower temperatures. Moreover, the peak area in the low-temperature region for LAFO-UR is larger than those for LFO and LAFO, and a new peak appears at 297 °C. This observation is likely since exsolution of Ag activates the oxygen species on the surface of LAFO. In order to reveal the reaction mechanism for CO catalytic oxidation, *in situ* Diffuse Reflexions Infrared Fourier Transformations Spectroscopy (DRIFTS) was examined (Fig. 4f). The peak positions of CO (at 2100–2200 cm⁻¹), HCOO (at 1560 cm⁻¹) and HCO₃^{*} (at 1437 cm⁻¹ and 1339 cm⁻¹) are observed, and the peak signal intensity of CO gradually decreases as the temperature increases, while the peaks for Ag adsorbing CO are not observed. Based on these observations, it can be deduced that with Ag as the absorption sites of O₂, the mechanism of CO oxidation conforms to the Eley–Rideal (E–R) mechanism. The presence of Ag weakens the Fe–O bonds,^{34,35} and the activated –OH on the surface of the catalyst reacts with CO to produce bicarbonate and formate, which subsequently react with O₂ adsorbed on the active site of Ag to produce CO₂. The reaction path of LAFO is the same as that of LAFO-UR (Fig. S15, ESI†).

In summary, an unconventional exsolution method was successfully applied to prepare a Ag_{0.03–x}/La_{0.87}Ag_xH_yFeO_{3–δ} composite for CO oxidation. The formation of Ag nanoparticles on the surface was confirmed by XPS and HRTEM. Hydrogen was found to be inserted into the perovskite lattice as indicated by FI-IR and TG-MS, which compensated for the cation losses caused by Ag exsolution and maintained the stability of the perovskite structure. The surface Ag nanoparticles increased the adsorption sites of O₂, weakened the Fe–O bond, and activated the lattice oxygen on the sample surface, resulting in superior CO oxidation activity. The ultrasonic reduction exsolution has demonstrated its potential for synthesizing efficient composite catalysts, holding great potential in the fields of energy, environment, and catalysis.

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Conflicts of interest

There are no conflicts to declare.

References

- D. Neagu, G. Tsekouras and D. N. Miller, *et al.*, *Nat. Chem.*, 2013, **5**, 916–923.
- A. Opitz, A. Nanning, C. Rameshan and R. Rameshan, *et al.*, *Angew. Chem., Int. Ed.*, 2015, **54**, 2628–2632.
- J. Zhu, S. Pang, T. Dittrich, Y. Gao and W. Nie, *et al.*, *Nano Lett.*, 2017, **17**, 6735–6741.
- Y. Jiang, Z. Geng, Y. Sun and X. Wang, *et al.*, *ACS Sustainable Chem. Eng.*, 2019, **8**, 302–310.
- D. Neagu, T. S. Oh, D. N. Miller and H. Ménard, *et al.*, *Nat. Commun.*, 2015, **6**, 8120.
- S. Wang, Y. Gao, S. Miao and T. Liu, *et al.*, *J. Am. Chem. Soc.*, 2017, **139**, 11771–11778.
- X. Wang, Z. Pan, X. Chu and K. Huang, *et al.*, *Angew. Chem., Int. Ed.*, 2019, **58**, 11720–11725.
- F. Lasserre, S. Suarez and F. Mücklich, *Mater. Res. Express*, 2016, **3**, 125007.
- M. Chu, C. Chen, Y. Wu, X. Yan and S. Jia, *et al.*, *Green Energy Environ.*, 2022, **4**, 792–798.
- Y. Cong, Z. Geng, Q. Zhu and H. Hou, *et al.*, *Angew. Chem., Int. Ed.*, 2021, **133**, 23568–23575.
- S. H. Cui, J. H. Li, X. W. Zhou and G. Y. Wang, *et al.*, *J. Mater. Chem. A*, 2013, **34**, 9689–9696.
- J. H. Myung, D. Neagu, D. N. Miller and J. T. Irvine, *Nature*, 2016, **537**, 528–531.
- Y. Gao, D. Chen, M. Saccoccio, Z. Lu and F. Ciucci, *Nano Energy*, 2016, **27**, 499–508.
- K. Zhu, T. Wu, M. Li, R. Lu and X. Zhu, *et al.*, *J. Mater. Chem. A*, 2017, **5**, 19836–19845.
- Y. Jin, W. Huo, L. Zhang and Y. Li, *et al.*, *Chem. Commun.*, 2021, **57**, 7168–7171.
- L. Yao, Z. Geng, W. Zhang and X. Wu, *et al.*, *ACS Sustainable Chem. Eng.*, 2020, **8**, 17194–17200.
- Y. Kim, H. S. Kim, D. Kang, M. Kim and J. W. Lee, *Chem. Eng. J.*, 2023, **468**, 143662.
- Z. Sun, W. Fan and Y. Bai, *et al.*, *ACS Appl. Mater. Interfaces*, 2021, **25**, 29755–29763.
- J. Ma, Z. Geng, Y. Jiang and X. Hou, *et al.*, *J. Alloys Compd.*, 2021, **854**, 157154.
- G. S. Pawar, A. Elikkotttil, S. Seetha and K. S. Reddy, *et al.*, *ACS Appl. Energy Mater.*, 2018, **1**, 3449–3456.
- B. Fu, J. Li, H. Jiang, X. He, Y. Ma and J. Wang, *et al.*, *Nanoscale*, 2022, **14**, 14073–14081.
- R. Li, Z. Wei, T. Huang and A. Yu, *Electrochim. Acta*, 2011, **56**, 6860–6865.
- L. Yao, Z. Geng, W. Zhang and X. Wu, *et al.*, *ACS Sustainable Chem. Eng.*, 2020, **8**, 17194–17200.
- F. Okejiri, Z. Zhang, J. Liu, M. Liu and S. Yang, *et al.*, *ChemSusChem*, 2020, **13**, 111–115.
- R. Huang, K. Kim and H. J. Kim, *et al.*, *ACS Appl. Nano Mater.*, 2019, **1**, 486–495.
- J. X. Liang, X. F. Yang, A. Wang and T. Zhang, *et al.*, *Catal. Sci. Technol.*, 2016, **18**, 6886–6892.
- L. Lv, Z. Li, Y. Ruan and Y. Chang, *et al.*, *Electrochim. Acta*, 2018, **286**, 172–178.
- P. Wang, N. W. Menzies, P. G. Dennis and J. Guo, *et al.*, *Environ. Sci. Technol.*, 2016, **15**, 8274–8281.
- Y. Cong, Q. Tang, X. Wang and M. Liu, *et al.*, *ACS Catal.*, 2019, **9**, 11743–11752.
- Y. Ma, N. Wassdahl, P. Skytt and J. Guo, *et al.*, *Phys. Rev. Lett.*, 1992, **17**, 2598–2601.
- H. Wang, L. Wu, J. Jiao and J. Zhou, *et al.*, *J. Mater. Chem. A*, 2015, **3**(12), 6517–6525.
- X. Wu, S. Xiong, J. Zhu, J. Wang, J. Shen and P. K. Chu, *Nano Lett.*, 2009, **9**, 4053–4060.
- X. Li, X. Wang, J. Ding and M. Ma, *et al.*, *ACS Catal.*, 2023, **13**, 6338–6350.
- H. Ha, S. Yoon, K. An and H. Y. Kim, *ACS Catal.*, 2018, **12**, 11491–11501.
- J. Zhang, L. Li, X. Huang and G. Li, *J. Mater. Chem.*, 2012, **22**, 10480–10487.

