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# Photoelectrocatalytic activity of silicon nanowires decorated with electroless copper nanoparticles and graphene oxide using a plasma jet for removal of methyl orange under visible light†

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Silicon nanowires (Si<sub>NWs</sub>) have been studied due to their interesting properties, such as light trapping and catalytic activity for removing organic molecules. In this work, silicon nanowires are decorated with copper (Si<sub>NWs</sub>-CuNPs), graphene oxide (Si<sub>NWs</sub>-GO), and both copper and graphene oxide GO (Si<sub>NWs</sub>-CuNPs-GO). They were prepared and tested as photoelectrocatalysts to remove the azoic dye methyl orange (MO). The silicon nanowires were synthesized by the MACE process using HF/AgNO<sub>3</sub> solution. The decoration with copper nanoparticles was made by galvanic displacement reaction utilizing a copper sulfate/HF solution, while decoration with GO was achieved using an atmospheric pressure plasma jet system (APPJ). The as-produced nanostructures were then characterized by SEM, XRD, XPS, and Raman spectroscopy. Cu(I) oxide was generated during the decoration with copper. Cu(II) oxide was produced when Si<sub>NWs</sub>-CuNPs were exposed to the APPJ. GO was successfully attached on the surface of silicon nanowires and silicon nanowires decorated with copper nanoparticles. The photoelectrocatalytic activity of silicon nanostructures was tested under visible light, leading to an MO removal efficiency of 96% within 175 min with Si<sub>NWs</sub>-CuNPs-GO, followed by Si<sub>NWs</sub>-CuNPs, Si<sub>NWs</sub>-GO, undecorated Si<sub>NWs</sub>, and bulk silicon.

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## 1. Introduction

In recent decades, the human population has increased abreast of its associated problems, such as environmental pollution, energy, and food necessities. Water contamination is a severe problem because the whole life on our planet depends on it. Wastewater is mainly produced by households and industries, such as pharmaceutical, paper, food processing, textile, hospitals, *etc.*<sup>1,2</sup> Dyes of synthetic origin have been extensively found in many of these industries and most of them are azo dyes. The presence of these types of compounds in wastewater can cause severe problems, including color and toxicity, and their intermediary products are associated with carcinogenic issues in human health.<sup>3</sup> Therefore, dye-based wastewater pollution is considered a serious problem and many researchers have focused on removing or degrading these pollutants by developing new technologies and materials.

Usually, physical, chemical, and biological processes are the conventional techniques for purifying wastewater containing dyes. However, these techniques are commonly expensive in their implementation and have a high environmental impact; for these reasons, oxidative processes, such as ozonation or chemical and photo-oxidation by the Fenton process, have been proposed for removing dyes.<sup>4,5</sup> Furthermore, among oxidative techniques, the electrochemical processes are attractive due to their facile control by current density and potential,<sup>6</sup> besides offering the possibility of combining with light irradiation to produce photoelectrochemical processes.

Silicon nanowires (Si<sub>NWs</sub>) have gained interest in the last few years due to their versatility and properties like absorption, light emission, catalytic activity, and a wide range of applications. Among all its applications, they can be exploited for photocatalysis,<sup>7-10</sup> solar water splitting,<sup>11,12</sup> detection of molecules,<sup>13-15</sup> light trapping,<sup>16,17</sup> reduction of metals,<sup>18</sup> storage and energy conversion,<sup>19</sup> CO<sub>2</sub> reduction,<sup>20,21</sup> hydrogen generation,<sup>22</sup> and many other applications. Moreover, it has been reported that the decoration of silicon nanowires with metallic nanoparticles can enhance their catalytic properties, especially for electron discharging. Copper is an economical, affordable non-noble metal that is very facile to put on the tips of silicon nanowires. Silicon nanowires decorated with copper nanoparticles (Si<sub>NWs</sub>-CuNPs) had been applied for coupling reactions

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of benzene,<sup>23,24</sup> electrochemical sensors,<sup>25</sup> photodiodes,<sup>26</sup> reduction of organic molecules such as 4-nitrophenol,<sup>27,28</sup> or the oxidation of organic pollutants.<sup>29</sup>

The versatility of silicon nanowires allows decoration with metallic nanoparticles and many other substances. For example, graphene oxide (GO) can be decorated on silicon nanowires in the investigation of their performance as detectors,<sup>30,31</sup> solar cells,<sup>32</sup> hydrogen evolution reaction,<sup>33</sup> or CO<sub>2</sub> reduction reaction.<sup>34</sup> In the field of pollutant remediation, the decoration of silicon nanowires with both copper and graphene oxide has been applied to remove ciprofloxacin.<sup>35</sup>

Among the methods and techniques used for the decoration of silicon nanowires with GO, spin-coating method,<sup>36</sup> chemical absorption,<sup>37</sup> electrophoretic deposition,<sup>35</sup> drop-casting method,<sup>30,31,33,34,38</sup> can be mentioned these are the most popular techniques for transferring GO to the silicon nanowire substrate. However, the atmospheric pressure plasma jet (APPJ) is another technique capable of controlling the treatment of several substrates. Plasma systems have been investigated because they are a powerful tool for the thermal treatment of sensitive materials such as plastics or glass substrates<sup>39</sup> and are also used in water technologies, biological materials, and agriculture.<sup>40,41</sup> The plasma technique is remarkable for being a low-operating, rapid, and non-polluting process, besides the facile integration to production lines and their use in flat samples as three-dimensional ones.<sup>42</sup> The plasma consists of reactive oxygen and nitrogen species, charged particles, molecules, free radicals, ions, atoms, and electrons that alter the hydrophobicity or hydrophilicity, roughness, or surface of the samples.<sup>41,43–45</sup>

The motivation for this work can be described as follows:

(i) The photoelectrocatalytic phenomenon of silicon nanowires is shown, wherein they can be used as a working electrode in visible light. Bulk silicon doesn't exhibit this phenomenon.

(ii) The novel use of an alternative coating method to drop-casting and electrophoretic deposition to obtain a homogeneous layer, covering the complete nanowire structure and not just the tip, thus increasing the contact area of the junction.

(iii) The Si<sub>NWs</sub> covered with GO could increase catalytic activity.

(iv) After plasma treatment for obtaining the GO coating, the silicon nanowires show photocatalytic or photoelectrocatalytic activity compared to untreated silicon nanowires.

In this work, we propose the modification of both Si<sub>NWs</sub> and Si<sub>NWs</sub>-Cu<sub>NPs</sub> with GO using an atmospheric pressure plasma jet (APPJ). The amount of GO deposited on the surface of the substrates can be controlled by varying the exposition time of the samples to the plasma flow. The catalytic activity of silicon nanowires decorated with GO was tested using methyl orange (MO) as the target molecule and compared against GO-unmodified silicon nanowires.

## 2. Experimental

### 2.1 Preparation of silicon nanowires (Si<sub>NWs</sub>)

Fabrication of silicon nanowires (Si<sub>NWs</sub>) was made by the metal-assisted chemical etching (MACE) method, which is followed by

our research group.<sup>46</sup> Briefly, p-type silicon wafers were cut into small pieces (2 × 3 cm) and cleaned with water, ethanol, and acetone. Then, the silicon plate was immersed in a HF dissolution (10% wt/wt) for 20 s, followed by immersion in a solution containing silver nitrate (0.35 M) and HF (5 M) for 10 s. Finally, silicon was etched by immersion of the piece into a solution containing H<sub>2</sub>O<sub>2</sub> (1.9 M)/HF (14.1 M) for 5 min at 58 °C. After that, the plate was rinsed with deionized water and dried naturally by the air.

### 2.2 Preparation of silicon nanowires decorated with copper (Si<sub>NWs</sub>-Cu<sub>NPs</sub>)

A fresh plate of Si<sub>NWs</sub> was immersed in a solution containing copper sulfate (0.1 M)/HF (0.1 M) for 5 s. Finally, the plate was rinsed with deionized water and dried in the atmospheric air.

### 2.3 Decoration with GO (Si<sub>NWs</sub>-GO and Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO)

Graphene oxide (GO) suspension was prepared by sonicating a solution containing 20 g L<sup>-1</sup> of GO for 3 h. The GO suspension was placed into an ultrasonic nebulizer, with the nozzle pointing directly to the silicon nanowires plate. GO was deposited using an atmospheric pressure plasma jet system (APPJ, Plasma Plus, Plasmatreteat FG5005S). The mist generated by the nebulizer was projected directly to the Si<sub>NWs</sub> and Si<sub>NWs</sub>-Cu<sub>NPs</sub> surfaces through the plasma stream, as is shown in Fig. 1. The deposition time, *i.e.*, the exposure time of the samples to the plasma stream with the mist was set to 2 min unless other time is indicated. In the same way, the deposition of GO was realized using small pieces of ITO and glass as blank substrates.

### 2.4 Photoelectrocatalytic setup

A three-electrode system was used for the photoelectrochemical experiments. A platinum wire was used as a counter electrode,

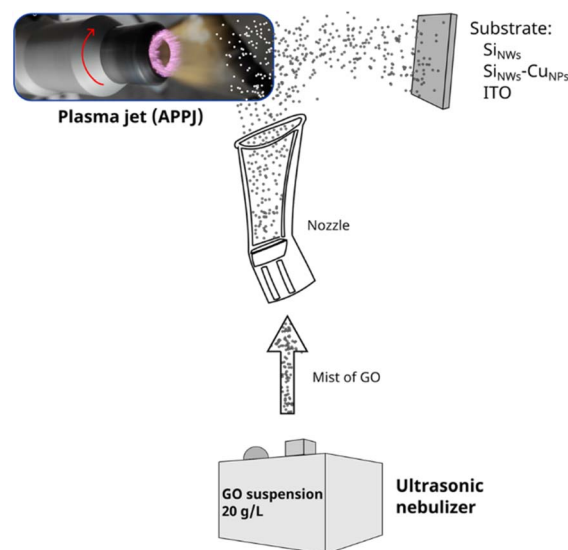


Fig. 1 Decoration of the substrates with GO using a plasma system (APPJ).



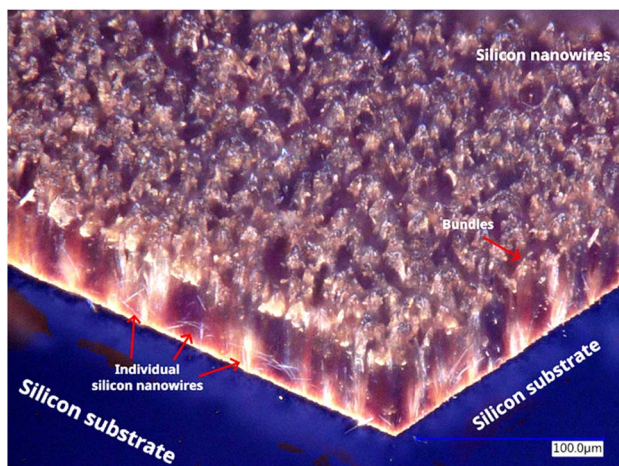


Fig. 2 Micrograph isometric view of silicon substrate containing Si<sub>NWs</sub> obtained by Digital Optical Microscopy.

while a saturated calomel was the reference electrode. Working electrodes were plates of Si<sub>NWs</sub>, Si<sub>NWs</sub>-Cu<sub>NPs</sub>, Si<sub>NWs</sub>-GO, and Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO, separately. A 25 mL of 20 ppm methyl orange (MO) solution containing 0.5 M K<sub>2</sub>SO<sub>4</sub> was placed in the cell as an electrolytic solution. The MO solution was purged with N<sub>2</sub> gas for 30 min to eliminate the remaining dissolved oxygen. Then, a current of 5 mA was applied to the system for 25 min. A sample of the solution was taken and analyzed by UV-vis

spectroscopy. The light source was a LED lamp with a power of 328 W m<sup>-2</sup>. In dark conditions, the lamp was removed. The evolution of methyl orange concentration was followed by UV-vis Shimadzu UV-2600 equipment. The control sample for light conditions was mounted as described above but omitting the presence of any electrodes. Concerning dark conditions, the control was similar to the light conditions but without the LED lamp or any catalysts. The removal percentage (*R*) of MO was calculated based on eqn (1):

$$R = \left(1 - \frac{C}{C_0}\right) \times 100\% \quad (1)$$

where, *C*<sub>0</sub> is the initial concentration of MO and *C* is the concentration of MO after the photoelectrocatalytic reaction.

## 2.5 Characterization of samples

Optical Digital Microscopy (Keyence VHX-5000x digital microscope) was used to observe the morphology of the obtained Si<sub>NWs</sub>. The morphology of all silicon nanowire samples was studied by Scanning Electron Microscopy (SEM). Energy Dispersive X-Ray Spectroscopy (EDS) was also used for the elemental analysis of all catalytic samples. SEM images and EDS analysis were obtained by JEOL JSM-6510LV scanning electron microscope. The crystalline structure of the samples was also determined by X-Ray diffraction (XRD) using a Bruker D8 Advance Diffractometer (Cu-Kα, λ = 1.5406 Å).

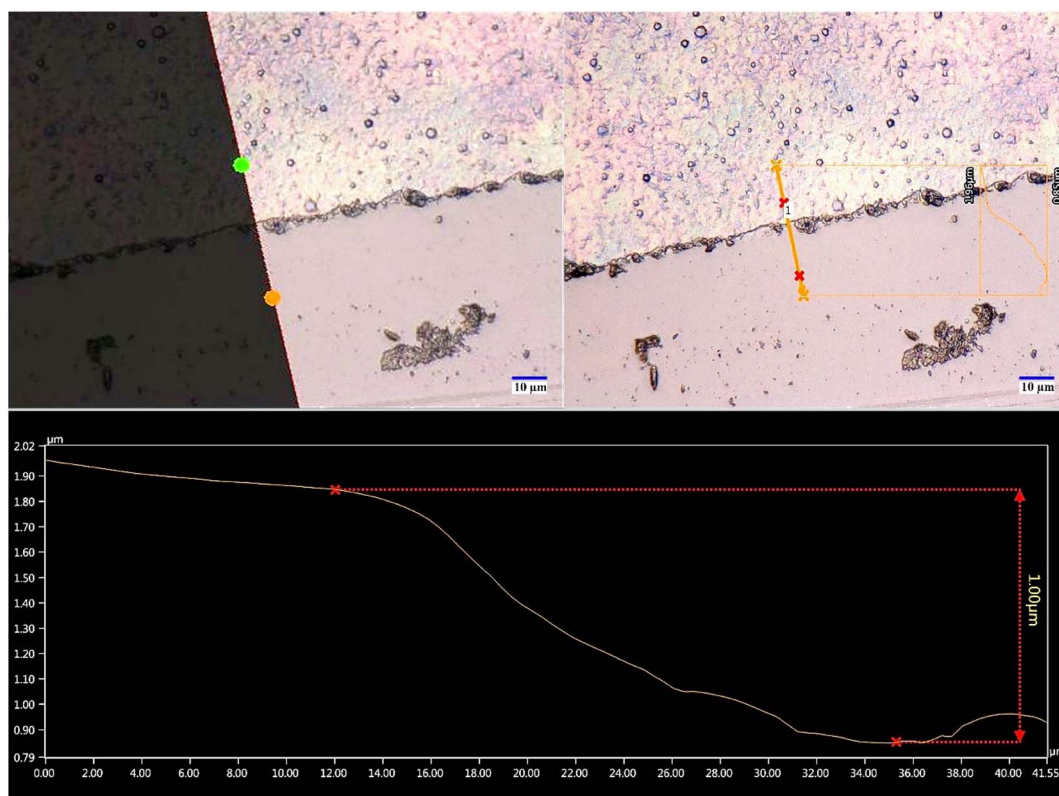


Fig. 3 Optical micrograph at 2000× of the GO coating on glass generated by APPJ using 2 min of deposition time. There is a continuous layer with iridescence caused by light interference in the coating. The profile shows a thickness of about 1 μm.



The samples were analyzed with a HORIBA LabRam micro-Raman spectrometer, model HR800. The laser line used was 532 nm with a nominal power of 73.9 mW.<sup>47</sup>

The composition of the samples was analyzed by XPS using a Thermo Scientific K-Alpha<sup>+</sup> spectrometer. The x-ray source line was Al K $\alpha$  ( $h\nu = 1486.6$  eV). The internal pressure in the chamber was about  $10^{-9}$  mBar and the spot size of about 400  $\mu\text{m}$ . The step energy was 20.0 eV, with ten scans and an initial erosion of 15 s to reduce absorbed atmospheric compounds.

## 3. Results and discussion

### 3.1 Characterization of silicon nanowires decorated with copper and GO

The obtained nanostructures were first observed by digital optical microscopy, where the morphology corresponds to silicon nanowire structures (Fig. 2). The cross-cut view of the silicon substrate containing Si<sub>NWs</sub> shows the vertically aligned structures and individual Si<sub>NWs</sub> at the edges of the plate. The average length of the obtained Si<sub>NWs</sub> was about 52  $\mu\text{m}$ . Agglomerations or bundles were also observed at the tips of the silicon nanowires, which can be associated with van der Waals forces during the drying process.<sup>48</sup> The micrograph also revealed a homogeneous distribution of the Si<sub>NWs</sub> on monocrystalline silicon. This means that the MACE process successfully allowed the production of vertically aligned silicon nanowires.

Fig. 3 shows an image of digital optical microscopy using 2000 $\times$  for a GO coating on a flat glass slide surface using a deposition time of 2 min. The coating is continuous, with characteristic topography and iridescence caused by light interference within the layer. In this case, the thickness was

about 1  $\mu\text{m}$ . The coatings deposited on the Si<sub>NWs</sub> were thinner since they had lower exposure time to the plasma jet. The GO coatings with 15 s, 30 s, and 60 s of deposition time could have thicknesses on flat surfaces of about 125 nm, 250 nm, and 500 nm, respectively.

The novelty of this coating yields the assembly of GO flakes of different sizes into a single continuous coating of graphitic structure. In addition, the plasma jet could unbind the hydrogen of the OH functional groups around the GO particles allowing them to bond with each other. This could be attained horizontally to cover the surface and vertically to increase the thickness of the coating.

Fig. 4 shows nine images of digital optical microscopy corresponding to three different views for three sets of samples with 15, 30, and 60 s of deposition time. The GO flakes in the aqueous solution mist went through the plasma jet to generate a single envelope that wrapped the Si<sub>NWs</sub> bundles. The cross-cut view, the top view, and the border between the Si<sub>NWs</sub> and crystalline silicon (c-Si) are shown in each row, respectively. The columns correspond to Si<sub>NWs</sub>-GO for the three deposition times.

The GO layer on the Si<sub>NWs</sub> was around the bundles and not directly on top of the structure. The thicker layers made the Si<sub>NWs</sub> structures darker but kept the open spaces. Access to inner surfaces was possible only in open spaces, not bundles. This was observed in the cross-cut views (Fig. 4 a, d and g).

Plates of Si<sub>NWs</sub>, Si<sub>NWs</sub>-Cu<sub>NPs</sub>, Si<sub>NWs</sub>-GO, and Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO were prepared and analyzed by SEM. Fig. 5 shows the as-obtained vertical structures, consistent with the results offered by other authors and interesting due to their capacity for light trapping, reducing light reflection, and catalytic activity.<sup>49–51</sup> As expected, the remaining silver can be observed at the bottom of silicon nanowires due to the MACE process

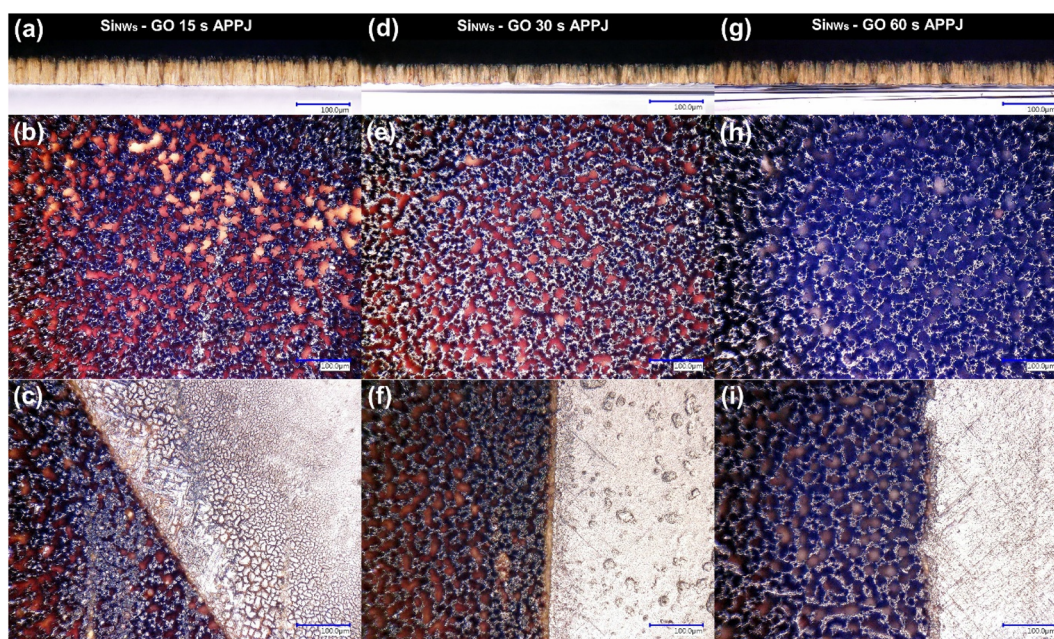


Fig. 4 Three different views for three sets of samples with 15 (a–c), 30 (d–f), and 60 (g–i) s of deposition time with the plasma jet for generating a GO envelope that wrapped the Si<sub>NWs</sub> bundles. The cross-cut view, the top view, and the border between the Si<sub>NWs</sub> and c-Si are shown in each row, respectively. The columns correspond to Si<sub>NWs</sub>-GO for the three deposition times.



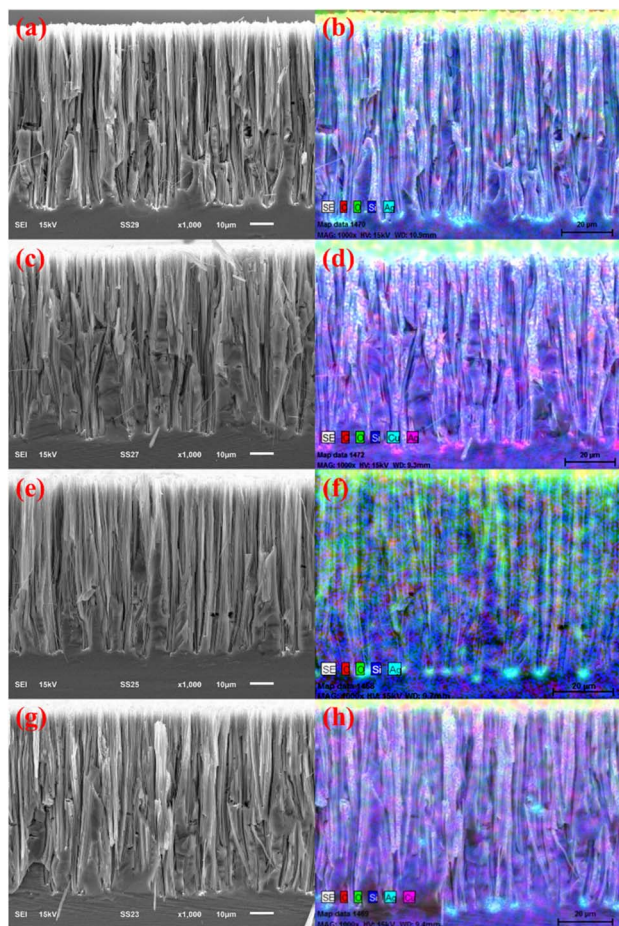


Fig. 5 SEM and EDS micrographs of (a)–(b)  $\text{Si}_{\text{NWs}}$ , (c)–(d)  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$ , (e)–(f)  $\text{Si}_{\text{NWs}}\text{-GO}$ , and (g)–(h)  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}\text{-GO}$ . The EDS mapping uses a color code for C (red), O (green), Si (blue), Ag (cyan), and Cu (pink), except for incise (d), where Ag and Cu interchanged colors.

(Fig. 5a–h). According to the elemental mapping of the samples (Fig. 5b, d, f, and h), the silver was located mainly at the bottom of each silicon nanowire sample. Porous structures along silicon nanowires and, in some cases, silver clusters were obtained and attributed to additional etching routes through secondary nucleation of silver nanoparticles.<sup>52</sup> Copper was located at the tip of  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$ , as expected (Fig. S3<sup>†</sup>), and oxygen was found along nanowires and at the tips, which may indicate the presence of copper oxide. Oxygen was also found distributed over the nanostructures, which is consistent readily with the oxidation of the silicon samples.

On the other hand, copper was scarcely found in the  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}\text{-GO}$  sample, probably due to the plasma flow exposure. Therefore, a sample of  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}\text{-GO}$  with 1 min of GO deposition time was prepared to compare the remaining copper on the samples (Fig. S6<sup>†</sup>). Comparing samples in which copper was added ( $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$ ,  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}\text{-GO}$  with 1 min of GO deposition time, and  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}\text{-GO}$  with 2 min of GO deposition time), the amount of copper was lower in the samples containing GO. According to the semi-quantitative analysis shown in Fig. S3, S5, and S6,<sup>†</sup> the weight percentage of copper

decreased from 1.28% in the  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$  sample to 0.87% in the  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}\text{-GO}$  sample for 1 min deposition time in comparison to a lower amount of 0.38% in the  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}\text{-GO}$  sample with 2 min deposition time. During the deposition of GO, the air/plasma flow probably dragged out most of the copper nanoparticles attached at the tips of silicon nanowires. Such  $\text{Cu}_{\text{NPs}}$  detachment was not previously seen in any test or treatment. However, some deposited copper nanoparticles could remain attached to the nanowires, which could be enough to exhibit catalytic activity. This means that the exposure time to the plasma flow could be a critical parameter for the quantity of material deposited and the composition of the original sample.

Samples of c-Si (*i.e.*, a silicon plate without nanowires or treatment),  $\text{Si}_{\text{NWs}}$ , and  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$  were analyzed by X-ray diffraction technique (Fig. 6). All the samples showed the silicon response for (111) planes, which has been observed in vertical silicon nanowires.<sup>48</sup> Furthermore, silver peaks were detected in both  $\text{Si}_{\text{NWs}}$  and  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$  samples. This implies that metallic nanoparticles are located at the bottom of nanostructures, as seen in SEM. Regarding  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$ , copper and copper(i) oxide peaks were detected in this sample. This confirms the correct attachment of copper nanoparticles by the experimental method used. Furthermore, the peaks of copper oxide in Fig. 6 indicate the presence of copper oxide besides the metallic copper nanoparticles. Hence, copper nanoparticles and copper(i) oxide were obtained at the tips of the  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$  samples.

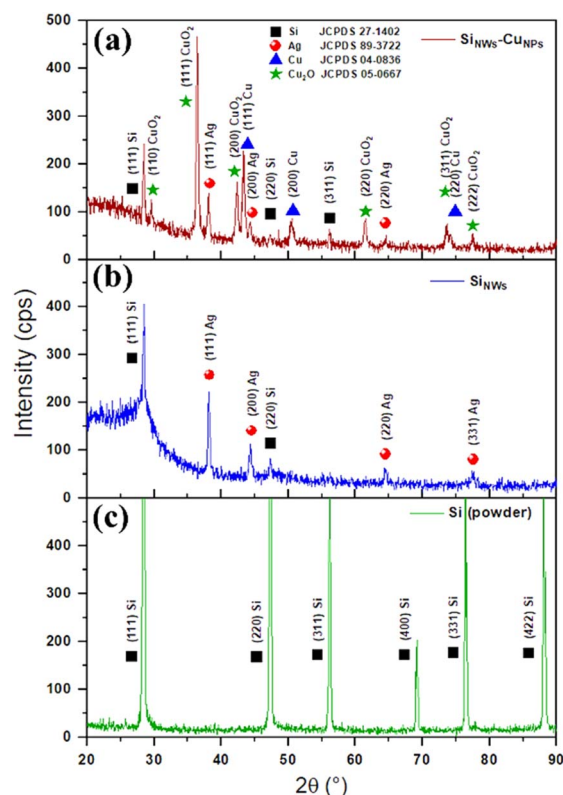


Fig. 6 XRD analyses for (a)  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$ , (b)  $\text{Si}_{\text{NWs}}$ , and (c) c-Si samples.



### 3.2 Electrochemical response of deposited GO on ITO

Deposited GO on ITO was first examined by cyclic voltammetry (Fig. 7). Compared with the bare ITO plate, the typical response of carbon materials was obtained. In the first cycle of the anodic direction, there were no peaks of any oxidation process in the window 0.5–1.0 V. This is expected for an oxidized material such as GO. As the cycle goes in the cathodic direction, reduction processes occur and oxidation peaks appear while returning to the anodic direction (peak i). During the cyclic scanning, the

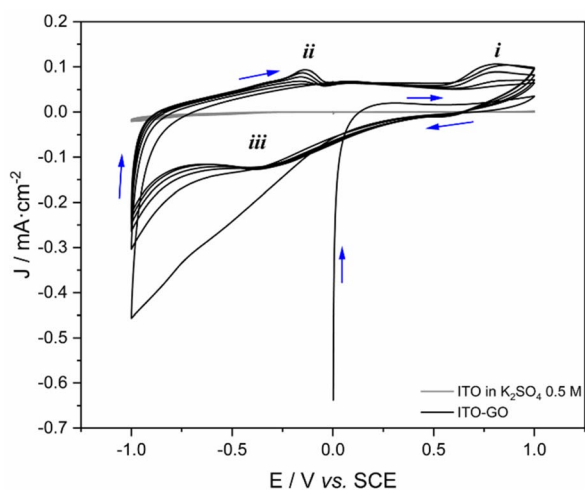


Fig. 7 Cyclic voltammetry of GO deposited on ITO surface. Arrows indicate the direction of cycles. Supporting electrolyte:  $\text{K}_2\text{SO}_4$  0.5 M;  $\nu = 20 \text{ mV s}^{-1}$ .

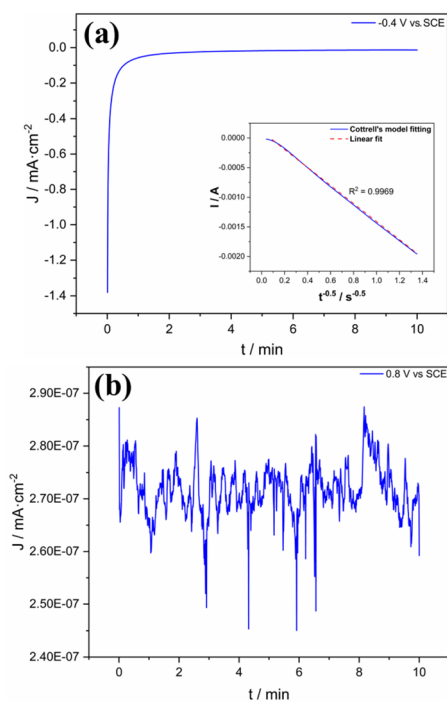


Fig. 8 Chronoamperometric measurements of GO deposited on ITO at (a)  $-0.4 \text{ V}$  and (b)  $0.8 \text{ V}$  vs. SCE during 10 min; inset: data adjusted to Cottrell's model. The supporting electrolyte was  $\text{K}_2\text{SO}_4$  0.5 M.

evolution of a redox pair (peaks ii and iii) was observed, which is associated with quinone/hydroquinone conversion.<sup>53,54</sup> The peak i has been reported in the graphene production from graphitic materials. It is related to the intercalation of ions containing sulfur functional groups,<sup>55,56</sup> which could be associated with sulfate ions from sulfuric acid and potassium sulfate.<sup>57,58</sup>

The nature of the deposited carbon material was also investigated through the chronoamperometry technique. As shown in Fig. 7, there was no oxidation peak at the potential 0.8 V in the first cycle. On the other hand, from the second cycle, there was a notable formation of peak iii. For this reason, the chronoamperometric experiments were performed by applying 0.8 and  $-0.4 \text{ V}$  vs. SCE for 10 min in two different fresh samples of ITO covered with GO (deposition time: 2 min, Fig. 8). The observed evolution of the current while applying  $-0.4 \text{ V}$  (Fig. 8a) can be associated to the peak iii and fitted to the Cottrell's model. This means that there is a process controlled by diffusion and, therefore, can be related to an electrochemically reducible material on the surface of ITO.

Regarding the imposition of 0.8 V (Fig. 8b), no significant current was obtained in this case. This behavior was maintained during the whole time of the imposition of potential. It fits with the observation at the same potential for the first cycle in Fig. 7. The fresh ITO-GO sample possesses an oxidized material, which can no longer be oxidized at 0.8 V without the imposition of a reduction process. The presence of oxidized species can explain the very small current obtained. Therefore, the evidence indicates the presence of oxidized carbon material on the surface of ITO and can be associated to GO.

### 3.3 Electrochemical behavior of silicon nanowires decorated with copper and GO

The surfaces of  $\text{Si}_{\text{NWs}}$  and  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$  decorated with GO were investigated electrochemically and compared with their bare counterparts (Fig. 9). The electrochemical response for  $\text{Si}_{\text{NWs}}$ -

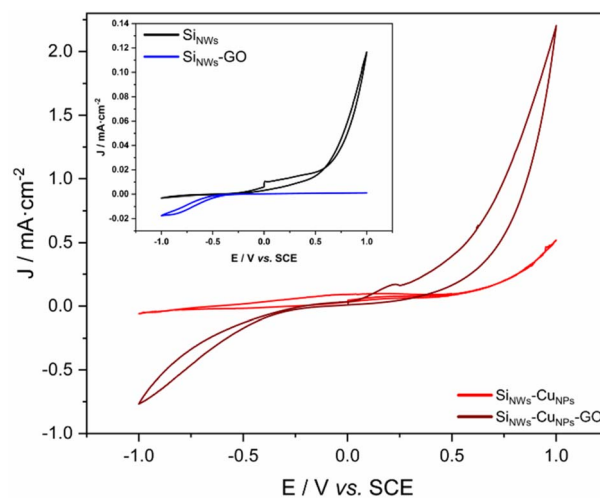


Fig. 9 Cyclic voltammetry of  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$  and  $\text{Si}_{\text{NWs}}\text{-Cu}_{\text{NPs}}$  decorated with GO. Inset: cyclic voltammogram of  $\text{Si}_{\text{NWs}}$  and  $\text{Si}_{\text{NWs}}\text{-GO}$ . Supporting electrolyte:  $\text{K}_2\text{SO}_4$  0.5 M;  $\nu = 20 \text{ mV s}^{-1}$ .



Cu<sub>NPs</sub>-GO was higher compared to Si<sub>NWs</sub>-Cu<sub>NPs</sub>. The increased current was also obtained in the cathodic window  $-1.0$  to  $-0.25$  V, similar to the response obtained with ITO-GO (Fig. 7). Regarding Si<sub>NWs</sub>-GO (Fig. 9, inset), the response was also higher as compared to Si<sub>NWs</sub>. In the anodic window of  $0.0$ – $1.0$  V, an oxidation current was observed in Si<sub>NWs</sub>, but only a capacitive current was found on Si<sub>NWs</sub>-GO. This effect can be due to the coverage of silicon nanowires with GO.<sup>37</sup> In the cathodic window of  $-1.0$  to  $-0.25$  V, the response of GO in Si<sub>NWs</sub>-GO was obtained and only capacitive current for Si<sub>NWs</sub> was present. The improvement of the current response of Si<sub>NWs</sub>-Cu<sub>NPs</sub> can be attributed to the presence of the more conductive copper, while the enhancement in Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO can be associated with the particles of GO.<sup>59,60</sup>

### 3.4 Characterization of silicon nanowires after electrochemical tests

Samples of Si<sub>NWs</sub>-GO and Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO were analyzed by XPS (Fig. 10a–g). An additional Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO sample was treated with several cyclic scans to obtain the electrochemical response of GO over the Si<sub>NWs</sub>-Cu<sub>NPs</sub>. After the treatment, the remaining sample was also analyzed by XPS (Fig. 10h–k). Deconvolution of

the signals of the samples was done using the Gaussian model. All the samples showed the presence of C–C/C–H bonds (284.7, 284.8 eV) as well as C–O–H/C–OH groups (286.51, 286.23 eV), indicating the presence of typical functional groups of GO.<sup>61</sup>

The signals for C–F bonds (293.18, 296.02 eV) were also found,<sup>62</sup> which can be related to the silicon nanowires etching process due to the solutions containing HF. Besides, oxygen deconvolution corresponds to the C–O–C bond (532.86, 532.5 eV). Also, the Cu–O bond (531.1 eV) was found in the Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO sample. In the Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO sample, after cyclic scans, signals were found for the C–O–C bond (532.66 eV) and C(O)O (533.08 eV). This is also in agreement with the Cu<sup>2+</sup> signals deconvoluted for both Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO (933.18, 953.1 eV) and Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO after cyclic scans (933.31 eV, 953.18 eV) of the samples.<sup>63,64</sup> The presence of copper(II) oxide is consistent with the production of this oxide during the exposition of the samples to the plasma flow<sup>39,65</sup> for the deposition of GO. No other significant differences were found between the samples containing copper. These results indicate the presence of oxygen-carbon functional groups associated with GO in the samples and copper(II) oxide in the Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO sample, as was also found by X-Ray diffraction analysis.

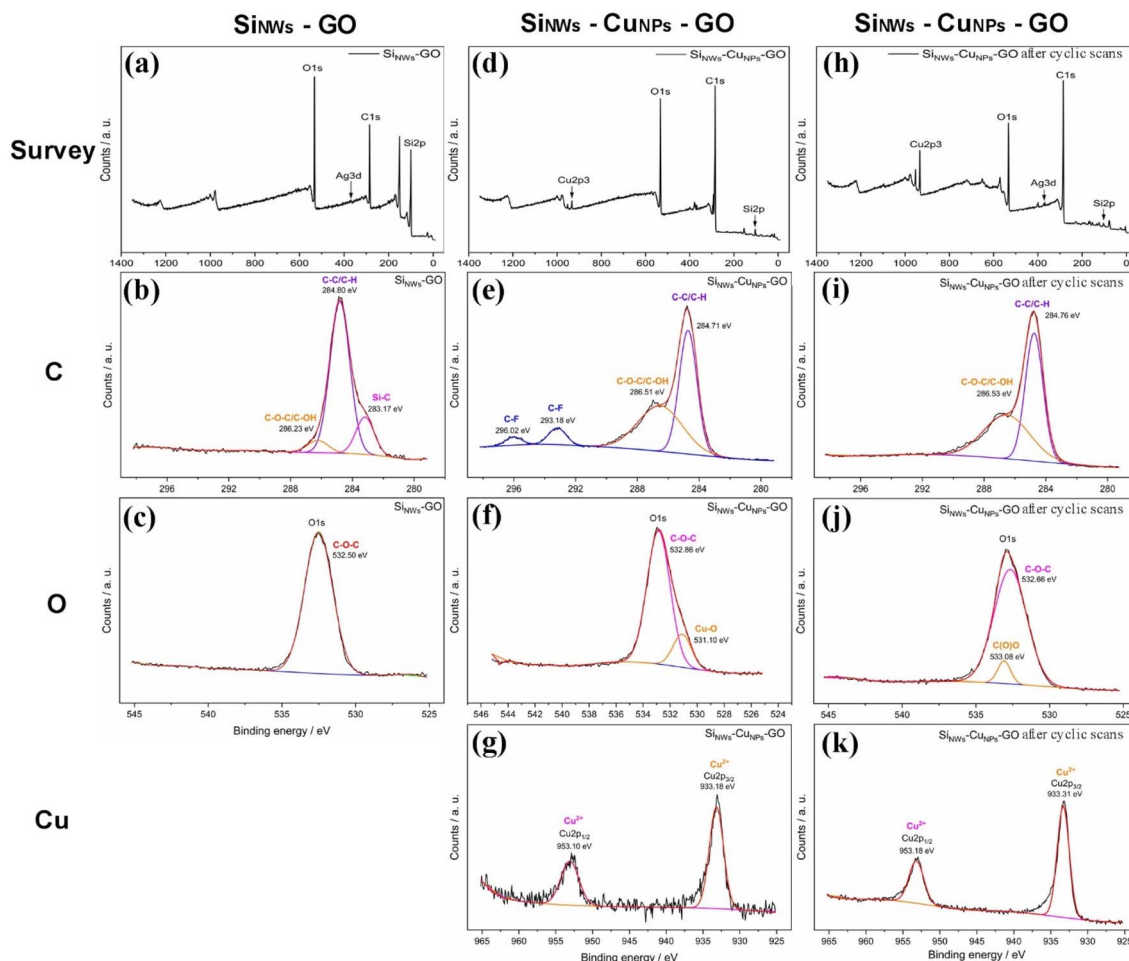


Fig. 10 XPS analysis of samples (a) Si<sub>NWs</sub>-GO, (d) Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO, and (h) Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO after cyclic scans (surveys); (b, e and i) carbon deconvolution for the samples; (c, f and j) oxygen deconvolution; (g and k) copper deconvolution.



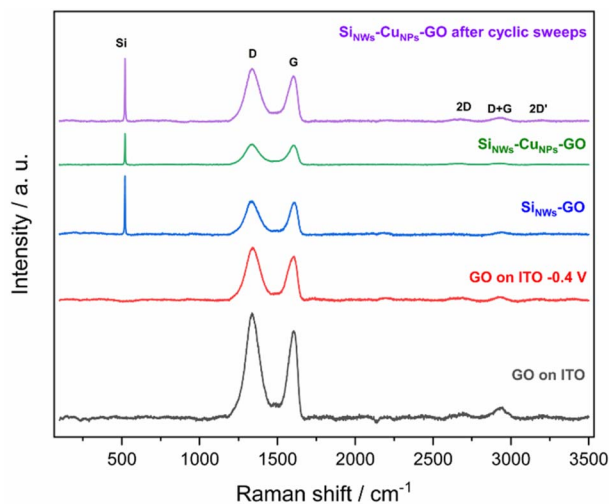


Fig. 11 Raman spectra of ITO covered with GO,  $\text{Si}_{\text{NWs}}\text{-GO}$ ,  $\text{Si}_{\text{NWs}}\text{-CuNPs-GO}$ , and  $\text{Si}_{\text{NWs}}\text{-CuNPs-GO}$  after cyclic scans.

Surfaces containing deposits of GO were studied by Raman analysis (Fig. 11). Deposits were obtained on the surface of ITO (deposition time: 2 min) compared to silicon nanowires substrate. A potential of  $-0.4$  V vs. SCE was applied in a GO sample over ITO to investigate the possible changes in the sample in the process of getting an electrochemically reduced sample of GO. In the same way, several cyclic scans were applied to a  $\text{Si}_{\text{NWs}}\text{-CuNPs-GO}$  sample to obtain the electrochemical response of GO over the  $\text{Si}_{\text{NWs}}\text{-CuNPs}$ . The samples were then analyzed by Raman spectroscopy to obtain evidence of the presence of GO. As is shown in Fig. 11, the presence of GO was confirmed in all the samples, and a peak corresponding to Si ( $520\text{ cm}^{-1}$ ) was detected in the silicon nanowires samples.<sup>66,67</sup> The D band, located at  $1319\text{--}1346\text{ cm}^{-1}$  corresponds to the presence of defects in the hexagonal structure of the graphene, while the G band (graphitic, located at  $1576\text{--}1580\text{ cm}^{-1}$ ) is associated with the  $\text{sp}^2$  carbon structures. The 2D band stands for the coherent double resonant process and is an indicator of the graphene layers in the sample.<sup>9,30,31</sup>

The D + G and 2D' bands were also found at  $2840$  and  $3120\text{ cm}^{-1}$ , respectively, which is a characteristic signature marker of GO.<sup>66</sup> A strong D peak and a moderate 2D band were observed in all cases, which implies defects, edges, or multilayer graphene structures.<sup>32</sup> Since the comparative sample (GO on ITO) showed defects within the GO structure, it is challenging to identify changes in the samples exposed to cyclic scans and  $-0.4$  V (the potential applied to produce reduced GO). Reduced GO is usually made in strong conditions, such as acidic media and using several cyclic scans,<sup>61</sup> which was not the case in this work. The ratio of intensities of D and G peaks ( $I_{\text{D}}/I_{\text{G}}$ ) is used as an indicator of defects in the deposits of GO. A higher ratio is expected for reduced GO than the original sample of GO due to the introduction of defects within the structure caused by the elimination of oxygen functional groups and the loss of carbon atoms.<sup>9,10,34,38</sup> Because the D peak was higher than G, it is not possible to conclude the presence of reduced GO in the samples

caused by the cyclic scans and applying  $-0.4$  V. After photoelectrocatalytic tests, as shown in the following section, the system did not have the conditions to reduce GO, neither the potential nor the time to accomplish significant reduction of GO. Nevertheless, the characteristic bands of GO on all the samples are enough to conclude its presence in the silicon nanowires and ITO. Hence, the successful deposition of GO was achieved.

### 3.5 Photoelectrocatalytic activity of silicon nanostructures

Photoelectrocatalytic activity of as-prepared  $\text{Si}_{\text{NWs}}$ ,  $\text{Si}_{\text{NWs}}\text{-CuNPs}$ ,  $\text{Si}_{\text{NWs}}\text{-GO}$ , and  $\text{Si}_{\text{NWs}}\text{-CuNPs-GO}$  have been probed using MO, promoting its remotion under visible light. Fig. 12a shows the typical UV-vis spectra of MO and its characteristic peak at  $465\text{ nm}$ .<sup>1,68</sup> The intensity of this peak was used for monitoring the remaining amount of MO in the solution over time. The continuous decrease in the MO peak intensity suggests the remotion of the dye over the surfaces of nanostructured catalysts. This behavior was observed in all silicon nanostructures used, but the degree of remotion found was different between them. The remotion of MO was also studied in the absence of light, as can be observed in Fig. 12c and d. The ratio of the initial concentration of MO ( $C_0$ ) to the concentration during the experiment ( $C$ )  $C/C_0$  was very low in the controls, which consisted of the solution without any catalyst. This implies the absence of self-degradation of MO under light and dark conditions.

In the presence of c-Si, the remotion of the dye was distinctive compared to the controls in both light and dark. In the absence of nanostructures, the remotion of the dye was poor, and due to photolysis, only 32.9% of the original dye in the solution could be removed. In general,  $\text{Si}_{\text{NWs}}$  exhibited a higher remotion capacity than the c-Si and the decorated  $\text{Si}_{\text{NWs}}$  with

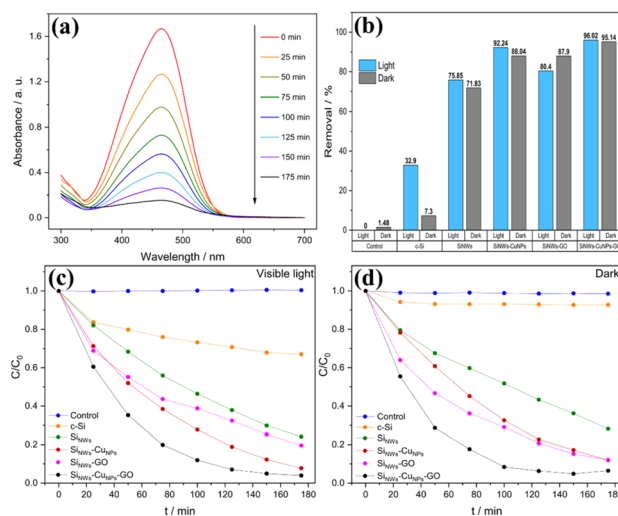


Fig. 12 (a) UV-vis spectra of MO samples during the experimentation time; (b) removal percentage of MO using the photocatalyst proposed in dark and light; (c) evolution of MO concentration over time in the presence of visible light and (d) dark using c-Si,  $\text{Si}_{\text{NWs}}$ ,  $\text{Si}_{\text{NWs}}\text{-CuNPs}$ ,  $\text{Si}_{\text{NWs}}\text{-GO}$ , and  $\text{Si}_{\text{NWs}}\text{-CuNPs-GO}$  as photocatalysts.





copper and/or GO showed higher removal capacity than bare Si<sub>NWs</sub>. The remotion obtained with Si<sub>NWs</sub> was 75.85% under light conditions compared to 71.83% in the dark, while the removal using Si<sub>NWs</sub>-Cu<sub>NPs</sub> was 92.24% in light and 88.04% in the dark (Fig. 10h). The enhanced remotion capacity can be explained by the presence of metallic nanoparticles on the surface of silicon nanowires (Cu<sub>NPs</sub>) that increases the electronic transfer and hence improve the catalytic activity during the photoelectrocatalytic process.<sup>69</sup>

The photocatalytic activity of semiconductor-metal catalyst mainly depends on the electron work function of the metal, and it is a significant parameter involved in the electronic transfer from the conduction band in the semiconductor to the metal.<sup>69,70</sup> In the presence of light, the remotion was higher than in the dark, as expected due to the photocatalytic properties of silicon nanowires, which capture light and promote the separation of electron-hole pairs.<sup>73</sup> Moreover, in the presence of GO, the nanostructures showed a contrary effect in Si<sub>NWs</sub>-GO. Although the remotion efficiency obtained with Si<sub>NWs</sub>-GO (80.4% in light, 87.9% in the dark) was superior to Si<sub>NWs</sub> in both light and dark conditions (75.85% in light, 71.83% in the dark), as shown in Fig. 12b, it was found that in the dark the remotion was higher than in light. This seems contradictory to the literature, which indicates that the GO on the surface of Si<sub>NWs</sub> can increase catalytic activity<sup>31-33,74</sup> besides light trapping and carrier generation. Regarding the Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO photocatalyst, a similar removal percentage was found in light and dark conditions (96.02% and 95.14%, respectively). These values are enhanced as compared to Si<sub>NWs</sub>-Cu<sub>NPs</sub> (92.24% in light and 88.04% in the dark). In general, the presence of copper nanoparticles led to more remotion capacity of Si<sub>NWs</sub>, and better dye discoloration was observed when decorating Si<sub>NWs</sub> with both copper nanoparticles and GO. Compared with data reported by other authors (Table 1), without considering specific test parameter values, a competitive removal capacity of Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO can be achieved within a few hours of photoelectrocatalytic activity.

The apparent rate constant also reflected the enhanced capacity of MO removal as promoted by copper and GO. The data of concentration ratio  $C/C_0$  was adjusted to the simplified Langmuir-Hinshelwood kinetics model as shown in eqn (2):<sup>75</sup>

$$\ln C = -k_r k_a t + \ln C_0 = -kt + \ln C_0 \quad (2)$$

where,  $k_r$  and  $k_a$  represent the constants for rate and adsorption, respectively, while  $k$  stands for the apparent rate constant. From the adjustment of data, the slope was related to the rate constant of each photocatalyst, as can be appreciated in Fig. 13a-f. In some cases, two linear ranges were found, which implies a change in the removal kinetics.

As expected, the lower values for rate constant were found with c-Si due to the absence of nanostructures capable of adsorbing and catalyzing the remotion of the dye. On the other hand, the rate constants found in Si<sub>NWs</sub> and Si<sub>NWs</sub>-Cu<sub>NPs</sub> photocatalysts were higher in the presence of light, in agreement with the removal percentage discussed above and with the photocatalytic properties of silicon nanowires. Nevertheless, the rate constant values for Si<sub>NWs</sub>-GO and Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO found were slightly higher in the dark. This is because the amount of GO deposited on the Si<sub>NWs</sub> surface can influence the performance of the catalytic activity of materials. The presence of GO makes the electron transfer faster between Si<sub>NWs</sub> and GO as well as Si<sub>NWs</sub>-Cu<sub>NPs</sub> and GO. However, when the amount of GO particles on the surface is large enough, the thickness of the GO layer and its associated series resistance can increase.<sup>36</sup> This can reduce the catalytic activity of the substrate and hence lower its removal capacity.

GO has valence and conduction band potentials of about 2.787 and -0.323 eV vs. NHE, respectively (Fig. 14).<sup>76,77</sup> These values are very near to the values of TiO<sub>2</sub>, which is a predominantly oxidizing photocatalyst, while Cu<sub>2</sub>O is a reducing one. The Si<sub>NWs</sub> structures are excellent in absorbing illumination and generating electron-hole pairs and they can be benefited by allocating Cu<sub>2</sub>O and GO or TiO<sub>2</sub>. In Z-scheme configurations, GO has also been applied as an electron receptor generated by the Si<sub>NWs</sub> by transferring them to the surface and protecting the Si<sub>NWs</sub>.<sup>9</sup>

Based on the evidence provided by the different photocatalysts employed in this work and, accordingly with the observations of other authors, the enhancement of catalytic activity is mainly related to electron-hole pair separation and faster carrier transport. When the different photocatalysts are irradiated with visible light while a current is applied to the

**Table 1** Removal efficiency values of dyes and pharmaceuticals using Si<sub>NWs</sub> with several decorating particles found in literature

Photocatalyst structure	Target molecule	Degradation or removal efficiency	Ref.
Si <sub>NWs</sub> -Ag <sub>NPs</sub>	Methylene blue	98–99% (200 min)	70
Hydrogen-terminated Si <sub>NWs</sub>	Methylene blue (4 mL, 10 <sup>-6</sup> M)	83.32% (200 min)	48
	Methyl orange (4 mL, 10 <sup>-6</sup> M)	93% (200 min)	
	Rhodamine B (3 mL, 5 × 10 <sup>-6</sup> M)	80% (240 min)	71
Si <sub>NWs</sub> -Au <sub>NPs</sub>	Rhodamine B (3 mL, 5 × 10 <sup>-6</sup> M)	80% (240 min)	71
Si <sub>NWs</sub> decorated with Pd	Methyl orange (260 mL, 25 mg L <sup>-1</sup> )	84.5% (microbial fuel cell, 36 h)	72
Si <sub>NWs</sub> -rGO	Indigo carmine (0.1 M)	> 99% (105 min)	37
Si <sub>NWs</sub> -rGO-CuO	Ciprofloxacin (4 mL, 10 mg L <sup>-1</sup> )	95.78% (in 0.6 mM PMS, 40 min)	10
Si <sub>NWs</sub> -GO	Methylene blue (200 mL, 6.4 mg L <sup>-1</sup> )	93% (280 min); 96% (in H <sub>2</sub> O <sub>2</sub> , 120 min)	9
Si <sub>NWs</sub> -Cu <sub>NPs</sub> -GO	Methyl orange (25 mL, 20 mg L <sup>-1</sup> , 175 min of photoelectrocatalysis)	96.02% in light, 95.14% in dark	This work
Si <sub>NWs</sub> -Cu <sub>NPs</sub>		92.24% in light, 88.04% in dark	
Si <sub>NWs</sub> -GO		80.4% in light, 87.9% in dark	
Si <sub>NWs</sub>		75.85% in light, 71.83% in dark	
c-Si		32.9% in light, 7.3% in dark	



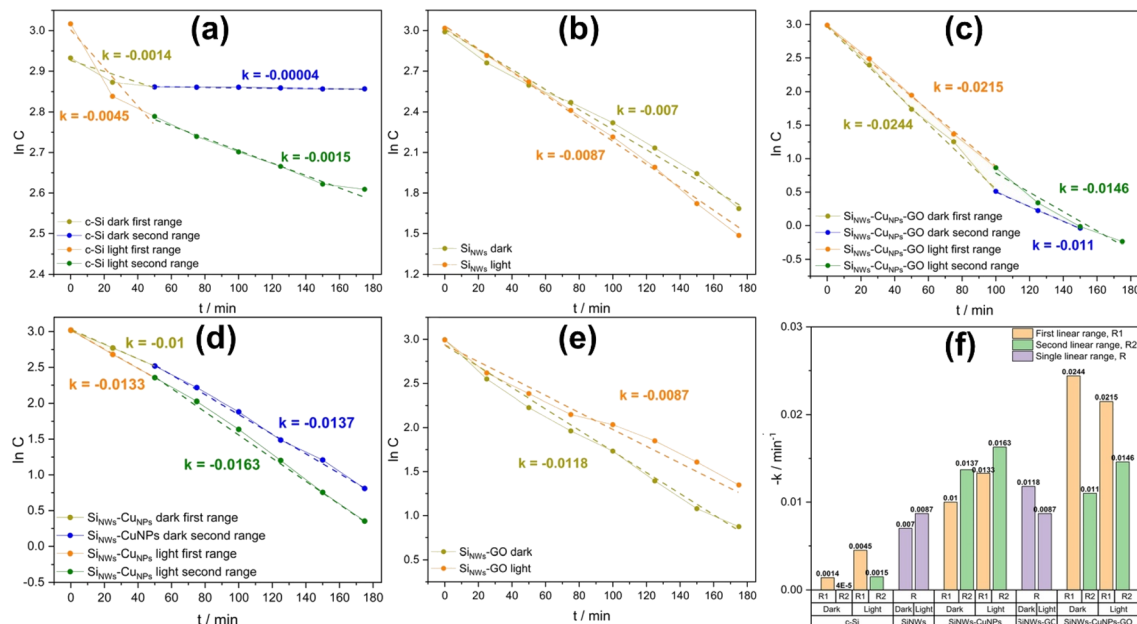


Fig. 13 Kinetics for MO removal using (a) c-Si, (b) SiNWs, (c) SiNWs-CuNPs-GO, (d) SiNW-CuNPs, (e) SiNWs-GO, and as photocatalysts. (f) Comparison for all rate constants found using the photocatalyst proposed.

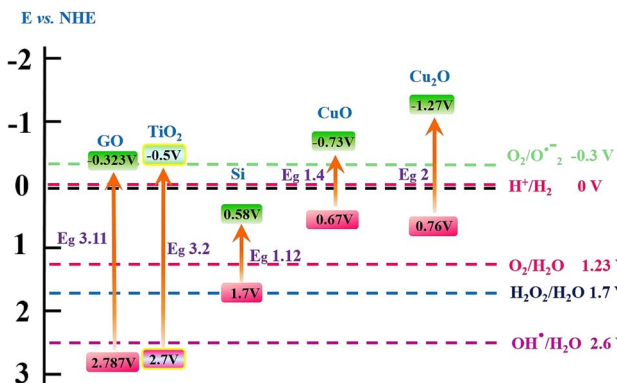


Fig. 14 Band gap, conduction band and valence band electrochemical potentials influence the mechanisms for making the photocatalytic processes efficient in the SiNWs, SiNWs-CuNPs, and SiNWs-CuNPs-GO. TiO<sub>2</sub> was included for comparison.

system, the photoexcitation produces the separation of electron-hole pairs. After the separation, some of the generated charge carriers are discharged on the surface of the SiNWs. In the case of SiNW-CuNPs, the generated electrons in the conduction band (CB) of SiNWs migrate to the CB of CuNPs while the holes are transported from the valence band (VB) of CuNPs to the VB of SiNWs and then to the anode. Similar dynamics of SiNWs-CuNPs can be expected for the SiNWs-GO catalyst, where the GO can facilitate the charge separation and transport electrons from SiNWs to GO. Furthermore, the Ag nanoparticles could promote electron discharge to generate reactive oxygen species contributing to dye bleaching.

Regarding the SiNWs-CuNPs-GO substrate, the generated electrons are transported from SiNWs through CuNPs and discharged to GO.<sup>75,78,79</sup> There has also been proposed a Z-scheme system for SiNWs-CuNPs-GO, where the photogenerated electrons

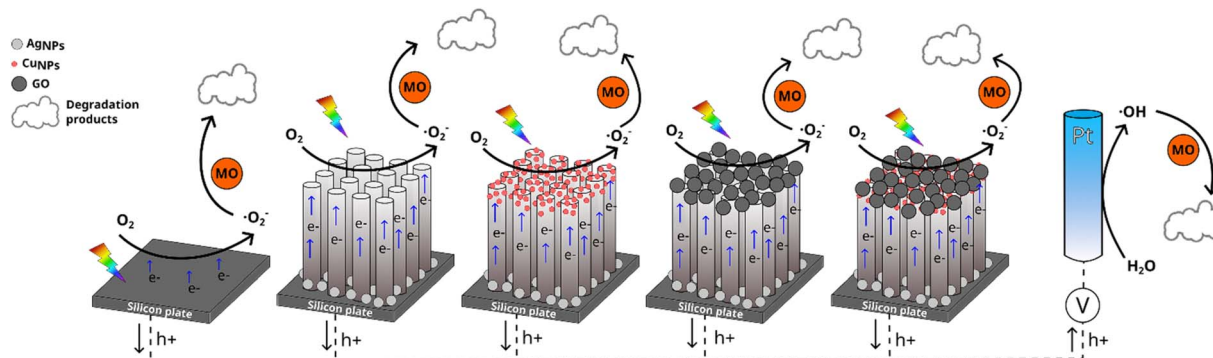


Fig. 15 Schematic illustration of the possible MO photoelectrocatalytic degradation/removal mechanism using c-Si, SiNWs, SiNWs-CuNPs, and SiNWs-CuNPs-GO. Dashed lines represent the possible connection to the platinum counter electrode.



in the CB of Cu<sub>NPs</sub> can react with the holes in the VB of Si<sub>NWs</sub> through GO while the electrons in the CB of Si<sub>NWs</sub> are transferred onto GO and then discharged to the electrolyte.<sup>10</sup> Once the carriers reach the photocatalyst–electrolyte interface, they can react with water and dissolved oxygen to produce reactive oxygen species (ROS). In addition, the ROS produced at the surface of the catalytic materials, such as ·OH and O<sub>2</sub><sup>·-</sup>, can lead to the oxidation of the dye. These processes are schematically represented in Fig. 15 for clear comprehension.

## 4. Conclusion

The Si<sub>NWs</sub> and Si<sub>NWs</sub>-Cu<sub>NPs</sub> were prepared by the metal-assisted chemical etching method (MACE). The EDS, XRD, and XPS analysis confirmed the composition of the photoelectrocatalysts, thus confirming the presence of silver at the bottom of the silicon nanowires and the copper and copper(I) oxide attached at the tips of the Si<sub>NWs</sub>-Cu<sub>NPs</sub>. In addition, copper(II) oxide was found on the samples exposed to the plasma.

The presence of GO on the surface of Si<sub>NWs</sub>-GO and Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO, as confirmed by Raman spectroscopy, corroborates that the samples are well covered with GO using the plasma deposition method.

The novelty of the GO coating is in the assembly of GO flakes of different sizes into a single continuous coating of graphitic structure. Furthermore, the plasma jet could unbind the hydrogen of the OH functional groups around the GO particles, subsequently allowing them to bond with each other. This could be attained horizontally to cover the surface and vertically to thicken the coating.

The heterostructures were tested using MO as a probe molecule, thus leading to its excellent removal (96.02% in 175 min) with Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO photocatalyst (light conditions), followed by the same photocatalyst in the dark (95.14%), Si<sub>NWs</sub>-Cu<sub>NPs</sub> (92.24% in light, 88.04% in the dark), Si<sub>NWs</sub>-GO (80.4% in light, 87.9% in the dark), Si<sub>NWs</sub> (75.85% in light, 71.83% in the dark), and c-Si (32.9% in light, 7.3% in the dark).

The enhanced photoelectrocatalytic properties of Si<sub>NWs</sub>-Cu<sub>NPs</sub>-GO were also evidenced in the rate constant, which was the highest compared with the other photocatalysts tested in this work.

The remotion efficiency obtained with Si<sub>NWs</sub>-GO (80.4% in light, 87.9% in the dark) was superior to Si<sub>NWs</sub> in both light and dark conditions (75.85% in light, 71.83% in the dark). However, the remotion was higher for Si<sub>NWs</sub>-GO in the dark than in light.

The proposed method for obtaining GO deposits is not limited to silicon substrate but to other surface kinds where the GO may improve the properties of the studied materials.

## Author contributions

Carlos Hernández Rodríguez: conceptualization, methodology, investigation, validation, supervision, writing–original draft, writing–review & editing. José de Jesús Pérez Bueno: conceptualization, methodology, investigation, validation, funding acquisition, supervision, writing–original draft, writing–review & editing. Alejandra Xochitl Maldonado Pérez: methodology,

assisted with experiments, investigation, writing–review & editing. Missael Ruiz Flores: methodology, investigation, assisted with experiments. Goldie Oza: conceptualization, methodology, investigation, validation, funding acquisition, supervision, writing–review & editing.

## Conflicts of interest

There are no conflicts to declare.

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