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Assessing the effect of surface states of mesoporous NiO films on charge transport and unveiling an unexpected light response phenomenon in tandem dye-sensitized solar cells†

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In this paper, the role of NiO surface states is assessed in a tandem dye-sensitized solar cell (t-DSSC) consisting of a 4-(Bis-(4-[5-(2,2-dicyano-vinyl)-thiophene-2-yl]-phenyl)-amino)-benzoic acid (P1) dye-sensitized NiO photocathode, a VG1-C8 dye-sensitized TiO₂ photoanode and the I⁻/I₃⁻ redox couple. The NiO surface states are proved to participate in the reduction of the I⁻/I₃⁻ electrolyte in the t-DSSCs. By adjusting the thickness of the TiO₂ film, the charge transport processes of the t-DSSCs are significantly affected by the photocurrent and the NiO surface states, resulting in various photovoltaic properties. This work also proves that the NiO surface states together with energy transfer between the desorbed P1 dye from the NiO photocathode and the VG1-C8 dye from the TiO₂ photoanode are responsible for the light response from both dyes observed in the IPCE spectra of the t-DSSCs.

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Introduction

Like most novel photovoltaic technologies, dye-sensitized solar cells (DSSCs) have experienced rapid progress since their breakthrough in 1991¹ and have attracted wide attention due to their ease of fabrication and high efficiency. Depending on the materials and assembly of the DSSCs, they can be divided into n-type, p-type, and tandem devices.² While the research on n-type DSSCs (n-DSSCs) has made good progress,³ the advances of p-type DSSCs (p-DSSCs) are still lagging behind that of the n-DSSC counterparts.^{4–8} This is especially an issue when assembling tandem DSSCs (t-DSSCs), which is a good alternative to surpass the Shockley-Queisser limit when compared with individual DSSCs.⁹ A traditional t-DSSC consists of a photoanode and a photocathode that share a redox couple. In theory, the tandem device that is connected in series should render a similar photocurrent density to its individual n-DSSCs and p-DSSCs if both of them have matched photocurrent density and optical absorption. The photovoltage of t-DSSCs theoretically is the sum of the photovoltage of the n- and p-DSSC parts, therefore enhancing the overall power conversion efficiency

(PCE) as compared to the individual n- or p-type devices.¹⁰ In addition, the concept of t-DSSCs has been used to build up dye-sensitized photoelectrochemical devices for solar fuel production.^{11,12}

Since 1999, when Lindquist and coworkers¹³ reported the first p-DSSC based on a dye-sensitized mesoporous NiO film, scientists have been working on the improvement of the performance of p-DSSCs and combining p-DSSCs with n-DSSCs to make t-DSSCs.^{14–19} Because of its large bandgap, large surface area, and good chemical-/thermo-/photo-stability, NiO is still the state-of-the-art p-type semiconductor used in p-DSSCs and t-DSSCs.²⁰ Many efforts have been made to improve the performance of NiO-based t-DSSCs.^{21,22} Traditionally, NiO films are regarded as a functional component for dye loading and the transport of holes that are injected by the excited dyes *via* its valence band (VB). According to previous reports, surface states of NiO are trap centers for charge recombination in p-DSSCs.^{23–27} In addition, there are also reports about the oxidation of I⁻ on NiO, indicating the charge recombination process between I⁻ and holes in the NiO VB.^{28,29} Recently, our group revealed that the surface states of NiO can also act as a catalyst for the reduction of I₃⁻ *via* electron transfer from the reduced dye.³⁰ The reduction of I₃⁻ by NiO surface states is considered a positive charge collection pathway rather than a charge recombination process. However, this work only studied the participation of NiO surface states in p-DSSCs in which no extra photocurrent generated from the counter electrode went through

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NiO film in the photocathode. However, in a t-DSSCs, the photoanode generates electrons that can directly reach the surface states of NiO through an external circuit. This previous work encourages us to elucidate the effect of the NiO surface states on the charge transport processes in t-DSSCs, consisting of a dye-sensitized TiO₂ photoanode and a dye-sensitized NiO photocathode.

In the present work, VG1-C8³¹ and P1³² dyes with complementary optical absorption spectra are used to sensitize TiO₂ and NiO films, acting as a photoanode and a photocathode, respectively. By regulating the thickness of the TiO₂ film, we can control the photocurrent from the TiO₂ photoanode, which allows us to study the effect of photoanode photocurrent on the performance and working mechanism of t-DSSCs. When the photocurrent densities of the photocathode and photoanode are similar, almost no extra electrons need to be transported *via* the surface states of the NiO film. This gives a standard t-DSSC, which shows approximately the same photocurrent density as that of the individual devices based on a photocathode or a photoanode, therefore rendering a higher fill factor. However, if the photocurrent density of the photoanode is higher than that of the photocathode, extra electrons generated from the photoanode, compared to the photocathode, can transport *via* the surface states of the NiO film and then reduce the species in redox couple, namely I₃⁻. Under this circumstance, the additional electrons are transported in or on NiO to reach the NiO surface states causing extra unwanted resistance. As a consequence, the fill factor of the t-DSSCs becomes unsatisfactory. Moreover, because the surface states of NiO can catalyze the reduction of I₃⁻, the photocurrent density of t-DSSCs is much higher than that of the individual p-DSSCs, which is different from our previous understanding of charge flow in a standard t-DSSC.

Results and discussion

The effect of NiO surface states on charge transport

Fig. 1a displays the molecular structures of VG1-C8 and P1 dyes that are used to sensitize the TiO₂ and NiO electrodes, respectively. As shown in Fig. 1b, VG1-C8 and P1 show complementary absorption spectra both in solution and on film, which matches the optical requirement of a t-DSSC. Fig. 2 shows the schematic buildup of the t-DSSCs with conventional working principles and corresponding energy levels of all of the components used in the device. The fabrication process of the t-DSSCs in this study can be found in the Experimental section and Scheme S1 in the ESI.† Traditionally, under light illumination, the dye on the photocathode is excited, and the excited dye injects a hole into the valence band (VB) of NiO to form a reduced dye that then reduces I₃⁻ of the I⁻/I₃⁻ redox couple. However, according to our previous study,³⁰ NiO surface states or an association species between the surface states and I₃⁻ can accept the electrons from the reduced dye. The catalytic behavior of the NiO surface states towards the I⁻/I₃⁻ redox couple could therefore change the working principle of the t-DSSCs from the traditional understanding.

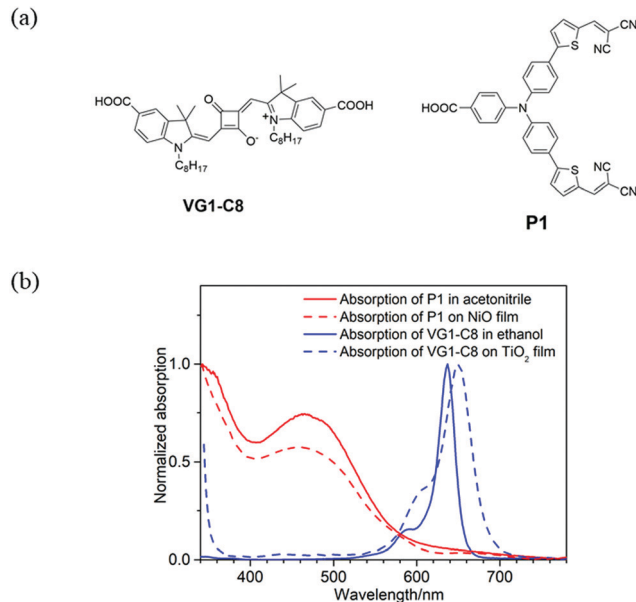


Fig. 1 (a) Molecular structures of VG1-C8 and P1 dyes and (b) normalized UV-Vis absorption spectra of VG1-C8 and P1 dyes in solution (extinction coefficient(ϵ): $\epsilon(\text{P1}) = 5.8 \times 10^4 \text{ M}^{-1} \text{ cm}^{-1}$ in acetonitrile,³¹ $\epsilon(\text{VG1-C8}) = 3.0 \times 10^5 \text{ M}^{-1} \text{ cm}^{-1}$ in ethanol).³²

To compare the catalytic performance of Pt and NiO films on reduction of I₃⁻, cyclic voltammetry (CV) was first employed. Fig. 3a shows the CV curve of Pt in 1,2-dimethyl-3-propylimidazolium iodide (DMPII)/I₂ (molar ratio: 2/1)-based electrolyte. An obvious reduction peak at a potential of $-0.4 \text{ V vs. Ag/AgNO}_3$ is observed, and it can be referred to as the reduction of I₃⁻ to I⁻. By replacing Pt with NiO (Fig. 3b), the reduction peak of I₃⁻ to I⁻ is shifted to a higher potential, $-0.65 \text{ V vs. Ag/AgNO}_3$. This suggests that NiO needs more overpotential to catalyze the reaction of I₃⁻ to I⁻ than Pt does. The results show that the NiO surface states are very important for catalytic activity. Without the surface states, the NiO film can't catalyze the reduction of I₃⁻.³⁰ In other words, Pt has a better electrocatalytic activity for I⁻/I₃⁻ redox couple reduction than the NiO film. Secondly, electrochemical impedance spectra (EIS) of the dummy cells (for Pt: Pt/electrolyte/Pt; for NiO: NiO/electrolyte/NiO) were also used to further evaluate the interface charge transfer resistance of Pt and NiO film with electrolytes. As shown in Fig. 4, the interfacial transfer resistance (R_{CE}) of Pt/electrolyte/Pt (Fig. 4a) is much lower than that of NiO/electrolyte/NiO (Fig. 4b), proving that Pt indeed catalyzes the reduction of I₃⁻^{33,34} more efficiently than the NiO film does. Note that the charge transport in or on NiO which is required to reach NiO surface states will certainly create an extra resistance, which adds an overpotential for the overall reduction of I₃⁻ to I⁻, as observed in CV and the large catalytic resistance in EIS. However, this does not necessarily mean that the surface states of NiO have intrinsically worse catalytic performance for I₃⁻ reduction when compared with Pt. Actually, as visible in the inset of EIS of NiO/electrolyte/NiO (inset in Fig. 4b), one can observe a small interfacial charge transfer resistance that appears in the high-frequency region, similar to the interfacial charge transfer resistance of the Pt electrode (Fig. 4a) which is only a little bit larger



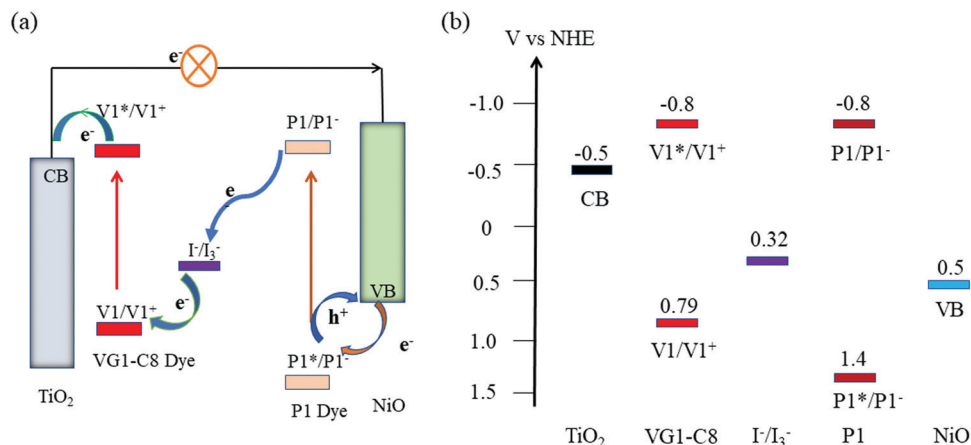


Fig. 2 (a) Schematic diagram of the conventional working principle of a t-DSSCs and (b) energy levels of all components involved in this work (NHE: normal hydrogen electrode; VB: valence band; CB: conduction band).

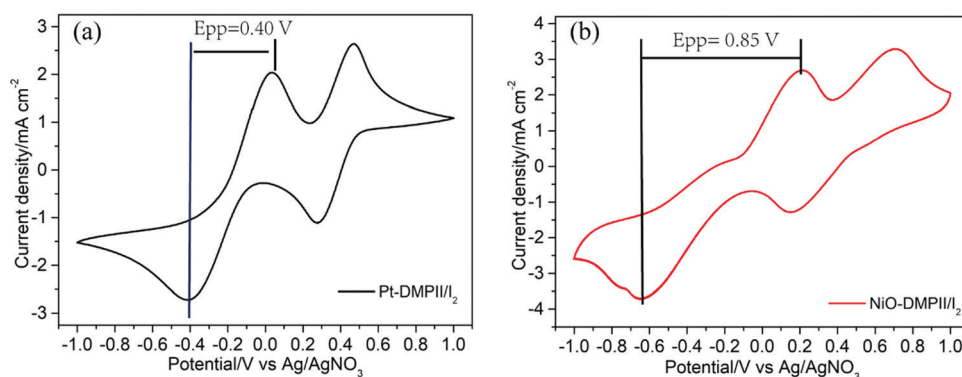


Fig. 3 Cyclic voltammetry of (a) a Pt electrode as working electrode and (b) a NiO film as working electrode in DMPII₂ electrolyte, scan rate: 100 mV s⁻¹.

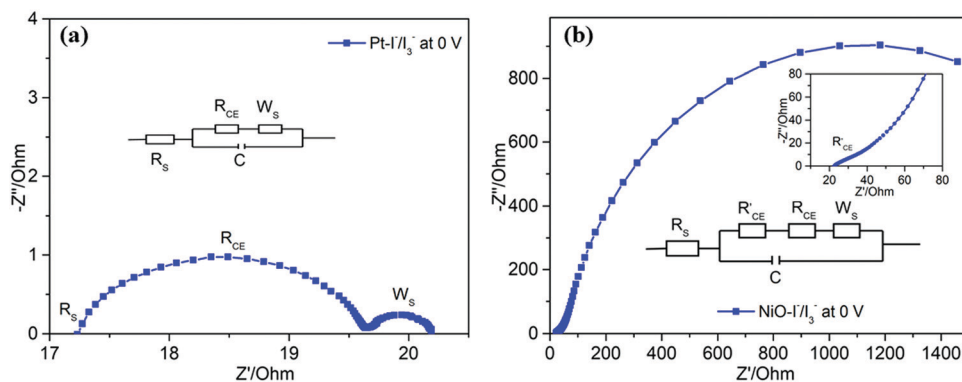


Fig. 4 EIS Nyquist plots of dummy devices of (a) Pt/electrolyte/Pt and (b) NiO/electrolyte/NiO (R_s refers to the series resistance from a conductive substrate; R_{CE} represents the resistance of charge transport in or on NiO film and back reaction, FTO/NiO; R_{CE} represents Pt or NiO/electrolyte resistance; W_s represents the electrolyte diffusion resistance).

than the Pt resistance. It is probably the real interfacial charge transfer resistance between the NiO surface states and the electrolyte. The large semicircle in the EIS of NiO/electrolyte/NiO then would correspond to the resistance of charge transport in or on the NiO surface and its charge recombination process.

To assess the effect of the NiO surface states on the photovoltaic performance, n-DSSCs, p-DSSCs, and t-DSSCs were assembled and evaluated. To make the structures of various n- and p-DSSCs devices in this paper easily distinguishable, we put the counter electrode in the brackets of the device name; for example, the device with the name TiO₂/VG1-C8 (Pt) means



that it is a solar cell with VG1-C8-sensitized TiO₂ as the photoanode and Pt as the counter electrode. For the t-DSSCs, the light illumination side is indicated in the bracket of the device name instead. For example, the Tandem device (n) represents that the t-DSSCs are illuminated from the n-type (photoanode) side.

Firstly, to match the photocurrent density of the TiO₂ photoanode with that of the NiO photocathode in the tandem device, a thin TiO₂ film with thickness of 2 μm was used. Fig. 5a displays the photocurrent density–photovoltage (*J*–*V*) curves of all devices, and the corresponding photovoltaic parameters are summarized in Table 1. For the n-DSSC TiO₂/VG1-C8 (Pt), a power conversion efficiency (PCE) of 0.466% was achieved with an open-circuit voltage (*V*_{oc}) of 0.626 V, a short-circuit photocurrent density (*J*_{sc}) of 1.00 mA cm⁻² and fill factor (FF) of 0.747. However, when the NiO film was employed to replace Pt as the counter electrode, the device TiO₂/VG1-C8 (NiO) rendered a PCE of 0.175% with a *V*_{oc} of 0.657 V, *J*_{sc} of 0.38 mA cm⁻² and FF of 0.728. *J*_{sc} of the TiO₂/VG1-C8 (NiO) device is much lower than for the TiO₂/VG1-C8 (Pt) device, which could be explained by extra resistance and lower catalytic performance of the NiO film for electrolyte reduction than Pt. This result is consistent with the previously discussed observations in CV and EIS. Notably, the FFs for both devices do not show a significant difference. This could be explained that not all surface states of NiO are involved when the photocurrent flow is low, meaning that the large resistance of the NiO film does not influence the FF. This is different from a device with a much higher photocurrent, which will be discussed in the following section on tandem devices with unmatched *J*_{sc} values between the photoanode and photocathode.

For the p-DSSC NiO/P1(Pt), a PCE of 0.024% with a *V*_{oc} of 0.065 V, *J*_{sc} of 1.09 mA cm⁻² and FF of 0.334 was obtained. With matching *J*_{sc} values between the TiO₂/VG1-C8 (Pt) and NiO/P1(Pt) devices, the t-DSSCs showed a PCE of 0.480% with a *V*_{oc} of 0.677 V, *J*_{sc} of 0.98 mA cm⁻² and FF of 0.723 when the solar cell was illuminated from the photoanode (Tandem device (n)). Due to the complementary absorption properties of the P1 and VG1-C8 dyes, the *J*_{sc} of the t-DSSCs was approximately the same as for the individual n-or p-DSSCs. Under illumination from the

Table 1 The photovoltaic parameters of DSSCs with a TiO₂ film thickness of 2 μm (the TiO₂ film is used in the n- and t-DSSCs)

Device	<i>V</i> _{oc} (V)	<i>J</i> _{sc} (mA cm ⁻²)	FF	PCE (%)
TiO ₂ /VG1-C8(Pt)	0.626	1.00	0.747	0.466
TiO ₂ /VG1-C8(NiO)	0.657	0.38	0.728	0.175
NiO/P1(Pt)	0.065	1.09	0.334	0.024
Tandem device (n)	0.677	0.98	0.723	0.480
Tandem device (p)	0.663	0.58	0.781	0.299

photocathode (Tandem device (p)), a PCE of 0.299% with a *V*_{oc} of 0.663 V, *J*_{sc} of 0.58 mA cm⁻² and FF of 0.781 was achieved.

To further probe the effect of the NiO surface states on the tandem DSSCs, a thicker TiO₂ film of 4.0 μm was used for the n- or t-DSSCs, and the corresponding *J*–*V* curves of the n-, p- and t-DSSCs are shown in Fig. 6a. Their photovoltaic parameters are summarized in Table 2. For the n-DSSC TiO₂/VG1-C8(Pt), a PCE of 1.250% with a *V*_{oc} of 0.624 V, a *J*_{sc} of 2.75 mA cm⁻² and a FF of 0.729 was achieved. When the NiO film was used to replace the Pt counter electrode, in the TiO₂/VG1-C8(NiO) device, it showed a PCE of 0.520% with a *V*_{oc} of 0.666 V, a *J*_{sc} of 1.53 mA cm⁻², and a FF of 0.509. Notably, the fill factor of the TiO₂/VG1-C8(NiO) solar cell is much lower than for TiO₂/VG1-C8(Pt). This result differs from the solar cells with a thinner TiO₂ film (2 μm) in which the FF values are similar in both solar cells. With increased TiO₂ thickness, more electrons are generated at the photoanode and flow to the photocathode. In order to compensate for all the electrons generated by the photoanode at the photocathode, more surface states of NiO film must be accessed and involved in the reduction reaction of I₃⁻ to I⁻. The electron transport in or on NiO film to reach more surface states will create a high resistance, as discussed in the CV and EIS sections, resulting in a low FF for the TiO₂/VG1-C8(NiO) device. For the p-DSSC NiO/P1(Pt), a PCE of 0.024% with a *V*_{oc} of 0.067 V, a *J*_{sc} of 1.05 mA cm⁻² and a FF of 0.338 was achieved. With a thicker TiO₂ film, the photoanode has a much higher *J*_{sc} than the photocathode. Therefore, it is interesting to see what happens if they are assembled in a tandem device and more electrons flow from the photoanode into the photocathode. For the t-DSSCs which are illuminated from the photoanode, a PCE of 0.580% with a *V*_{oc} of 0.656 V, a

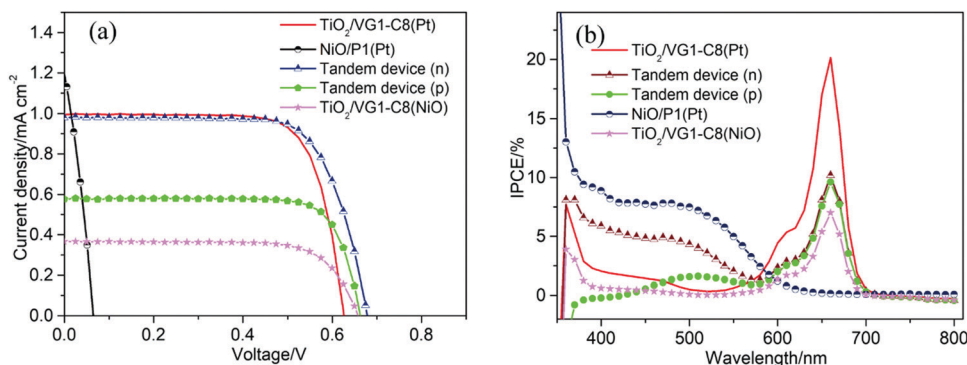


Fig. 5 (a) The *J*–*V* curves of solar cell devices based on TiO₂ film with thickness of 2 μm under AM 1.5G (Mask area: 0.16 cm²; electrolyte: I₂, DMPII and TBP with a molar ratio of 0.05 M : 0.6 M : 0.05 M in acetonitrile) and (b) IPCE spectra of solar cell devices.



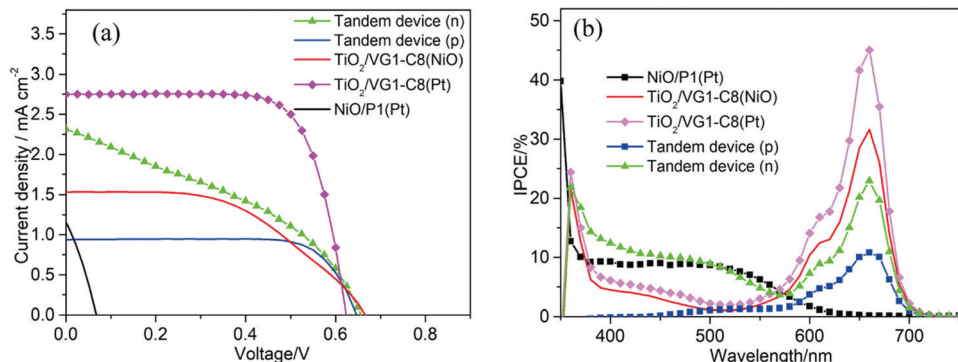


Fig. 6 (a) J - V curves and (b) IPCE spectra of all DSSCs based on $4 \mu\text{m}$ TiO_2 .

Table 2 Photovoltaic parameters of DSSCs with a TiO_2 film thickness of $4 \mu\text{m}$

Device	V_{oc} (V)	J_{sc} (mA cm^{-2})	FF	PCE (%)
$\text{TiO}_2/\text{VG1-C8(Pt)}$	0.624	2.75	0.729	1.250
$\text{TiO}_2/\text{VG1-C8(NiO)}$	0.666	1.53	0.509	0.520
NiO/P1(Pt)	0.067	1.05	0.338	0.024
Tandem device (n)	0.656	2.28	0.388	0.580
Tandem device (p)	0.647	0.94	0.758	0.461

J_{sc} of 2.28 mA cm^{-2} , and a FF of 0.388 was reached. This FF value is significantly lower than that of the tandem device with a thin TiO_2 film, 0.723 (with a matched photocurrent between the photocathode and the photoanode), which is caused by the need to transport more electrons in or on NiO film to reach the surface states of NiO for I_3^- reduction as more electrons flow from the photoanode into the photocathode. This result is therefore consistent with the observations for the $\text{TiO}_2/\text{VG1-C8(NiO)}$ device.

Based on the photovoltaic data, we can now conclude the roles of surface states of NiO film in t-DSSCs. As elucidated in Fig. 7, when the J_{sc} values of the photoanode and photocathode are matched, all or the majority of electrons from the photoanode can be supported by the injected holes in NiO from excited P1 dye (Process I). Then, the reduced P1 dyes give electrons to NiO surface states which are used for the reduction of I_3^- . As the injected holes in NiO are taken away by external electrons (e.g.: a bias potential), the charge recombination between electrons in the surface state and injected holes in NiO is significantly inhibited. Generally, the serious charge recombination in p-DSSCs is responsible for its low FF.³⁵ Therefore, for the tandem solar cells with matched photocurrent from both photoelectrodes, the FF of the device is improved as compared to that of the individual p-DSSCs.

However, when the J_{sc} of the photoanode is much larger than that of the photocathode, the holes in NiO VB are not sufficient to accommodate all the electrons coming from the photoanode and the NiO surface states start playing a role. In addition to process I, the extra electrons need to move in or on the NiO film to reach the NiO surface states and then reduce the redox couple (Process II) which therefore creates a larger transport and interface resistance. The consequence is a lower

FF for the tandem DSSCs with an unmatched J_{sc} from both photoelectrodes. According to the traditional understanding of the t-DSSCs in series, the J_{sc} of the tandem DSSCs should be equal to or lower than that of the photocathode and the photoanode. However, due to the role of NiO surface states, the J_{sc} of t-DSSCs with an unmatched photocurrent from both photoelectrodes is actually in the range of the J_{sc} of the individual p- and n-DSSCs, but much closer to that of the photoanode. As shown in Fig. 7, when the J_{sc} of the photoanode is much higher when compared with the photocathode in a tandem DSSC, the t-DSSCs should have two parallel charge transport processes involved: one is with NiO/P1 as a dark counter electrode, as in n-DSSCs ($\text{TiO}_2/\text{VG1-C8(NiO)}$), and the other is with NiO/P1 as a photoelectrode, as in an ideal t-DSSC. Dai and co-authors³⁶ reported a diode model to predict the J - V curve of the tandem DSSCs. In their results, the V_{oc} of tandem DSSCs is the sum of n-type and p-type. The current density of tandem DSSCs is determined by the photoanode, which is almost equal to the n-type device. However, in our case, the photocurrent density of tandem DSSCs is actually in the range of the J_{sc} of the individual p- and n-DSSCs, but much closer to that of the photoanode, when the photocurrent densities of the photoanode and photocathode are mismatched. The mentioned diode model seems only to fit well for the matched photocurrent density between the photoanode and photocathode. The proposed working principles of the tandem DSSCs in this work can also explain the phenomenon beyond the case that the photocurrent matches between two photoelectrodes.

Mysteries of IPCE spectra of tandem DSSCs

From the IPCE spectral response of individual n- and p-DSSCs (Fig. 5b and 6b), it is obvious that the P1 and VG1-C8 dyes have well-matched photoresponse spectra, with little overlap. According to the basic working principle of an ideal t-DSSC, one should not expect to see an IPCE response from either VG1-C8 or P1 dye, except for the overlapping absorption region between these two dyes where both dyes can be excited and generate electron flow in t-DSSCs. However, we did see this unexpected IPCE response from both VG1-C8 and P1 dyes in the tandem device. Similar to our observations, other groups



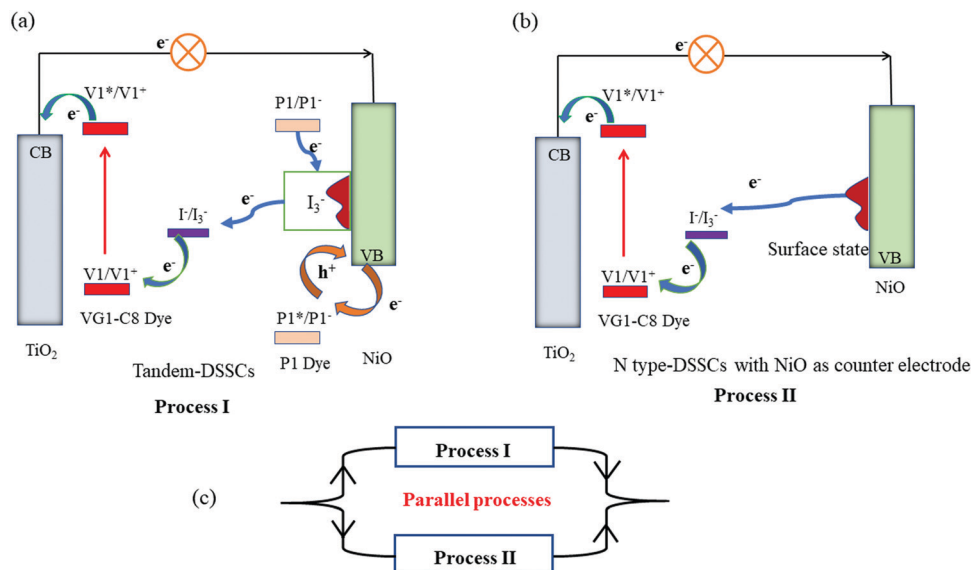


Fig. 7 Proposed working principles of the tandem DSSCs with NiO film as a photocathode in (a) a standard t-DSSC; (b) an n-DSSC with NiO as counter electrode; (c) parallel processes (process I and II) that are involved in a tandem device.

have also found IPCE responses from both dyes in the t-DSSCs.^{37–39} As discussed above, since the NiO surface states can catalyze the I_3^- reduction, the photo-response from VG1-C8 dyes in IPCE of the t-DSSCs can be reasonably explained by the NiO film working as a counter electrode. However, the photo-response of P1 (400–600 nm) in the IPCE spectra for t-DSSCs is mysterious, as TiO₂ cannot act as a counter electrode to catalyze the I^-/I_3^- electrolyte in a p-type DSSC,²⁸ which is evident from a solar cell NiO/P1(TiO₂) that did not show any P1 spectral response in the IPCE spectrum (Fig. S2, ESI[†]).

Using a P1-sensitized ZrO₂ film (P1/ZrO₂) instead of the P1/NiO film in the t-DSSCs, with Pt on the FTO surface to catalyze I_3^- reduction, the spectral response of P1 in the IPCE spectra of such t-DSSCs was still remaining (Fig. 8). This led us to consider electron and energy transfer between P1 and VG1-C8 dyes as a possible explanation. The distance between P1 on the photocathode and VG1-C8 on the photoanode is about 25 μm , which should preclude electron transfer and energy transfer directly between the dyes on the two photoelectrodes. However, if the P1 dye desorbs from the NiO surface, it

could diffuse to come in close proximity to the photoanode, which makes the electron and energy transfer possible. To check if the P1 dye desorbs from the photocathode in the presence of a solvent, a NiO/P1 film was immersed into acetonitrile/*tert*-butyl pyridine (same ratio as in the electrolyte) for 1 h to check its absorbance (Fig. S3, ESI[†]). The data show that the absorbance of the NiO/P1 film decreases from 0.4 to 0.2 at 480 nm. The result indicates that the P1 dye can indeed detach from NiO film in the presence of electrolyte solvent and come in contact with, and even attach to, the TiO₂ film. Electron transfer from the ground state VG1-C8 to excited P1 is thermodynamically favorable (Fig. 2b) and could be followed by electron injection from reduced P1 to TiO₂. Alternatively, the emission of P1 has a good overlap with the absorption of VG1-C8, so the process of energy transfer between desorbed P1 and VG1-C8 is clearly possible.

To verify that energy transfer between two dyes is possible, the fluorescence of a certain concentration of VG1-C8 acetonitrile solution was recorded in the presence of various amounts of P1 dye. As shown in Fig. 9a, an increasing addition of P1 dye

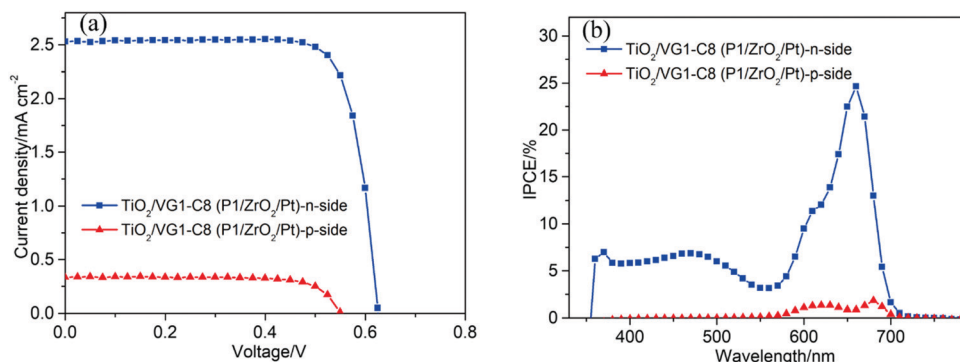


Fig. 8 (a) The J - V curves and (b) IPCE spectra of the TiO₂/VG1-C8 (P1/ZrO₂/Pt)-based device.



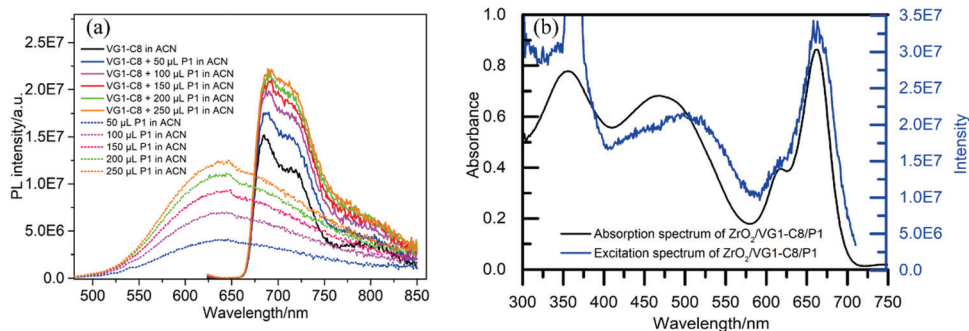


Fig. 9 (a) The PL spectra of P1 and VG1-C8 dye in acetonitrile (excitation at 480 nm). (b) The absorption and fluorescence excitation spectra (emission at 680 nm) of $\text{ZrO}_2/\text{VG1-C8/P1}$.

into the VG1-C8 solution (Fig. 9a) leads to the quenching of the intensity of P1 dye at 650 nm. In contrast, the PL intensity of VG1-C8 dye at 680 nm was increased, which suggests its sensitization by energy transfer from P1^* . In order to observe the energy transfer between two dyes on a film, the absorption spectrum and excitation spectrum monitoring the emission at 680 nm (Fig. 9b) of $\text{ZrO}_2/\text{VG1-C8/P1}$ film were measured. These two spectra gave similar shapes, with a strong contribution from the absorption bands of both dyes to the VG1-C8 fluorescence. Moreover, the emission spectrum of the $\text{ZrO}_2/\text{VG1-C8/P1}$ film (Fig. S4, ESI[†]) shows that the emission peak of the P1 dye is largely quenched. These results prove that there is indeed energy transfer from P1 to VG1-C8 dye both in solution and on film. Although the absorption of I_3^- extends to 500 nm, one can see that there is no IPCE response at 500 nm at all in the absence of the dye from the previous report.⁴⁰ If we considered that the IPCE response of the device without dye is from the electrolyte, then the spectral overlap between the emission of P1 and absorption I_3^- should not be possible. Kim and co-workers³⁷ used a cobalt-based redox couple in tandem DSSCs with P1 and SQ dyes. The IPCE response from both dyes can still be observed in the tandem DSSC. Then, we can exclude the Förster resonance energy transfer (FRET) between P1 and I_3^- . Based on the spectral overlap of the two dyes, the energy transfer between the two dyes should be from FRET.^{41,42}

To further make sure that the observed IPCE response from both dyes in t-DSSCs is indeed due to the FRET between the two dyes, a TiO_2 film was first dipped into the VG1-C8 solution for 12 h and then sensitized with P1 dye solution for 6 h. A $\text{TiO}_2/\text{VG1-C8/P1}$ (Pt) cell was fabricated (Table S4, ESI[†]). For such a cell, the IPCE spectrum in Fig. 10b shows a high IPCE value for the P1 dye. However, when a bare TiO_2 film was sensitized with the P1 dye ($\text{TiO}_2/\text{P1}$), the IPCE value P1 was much lower than that of the $\text{TiO}_2/\text{VG1-C8/P1}$ (Pt) device, probably due to poor electron injection from the excited P1 to TiO_2 . This result indicates that there is indeed FRET happening between P1 and VG1-C8. When the P1 dye (1 mg mL^{-1}) is introduced into the electrolyte of the $\text{TiO}_2/\text{VG1-C8}$ solar cell, the IPCE response of P1 is also observed. It can therefore be concluded that the spectral response of the P1 dye in the IPCE spectra of tandem solar cells is highly possible from FRET between the desorbed P1 dye and the VG1-C8 dye on the electrode, with the NiO film and its surface states working as a counter electrode. This result also explains why the P1 IPCE response in a tandem solar cell is weaker with illumination from the p-side than from the n-side (Fig. 5), since the remaining P1 dyes on the photocathode filter the light. Although these experiments still do not completely exclude another pathway: electron transfer from VG1-C8 to excited P1, followed by electron injection from P1-, the experimental results do suggest inefficient injection from P1^* which gives time for FRET to happen. This implies that the FRET

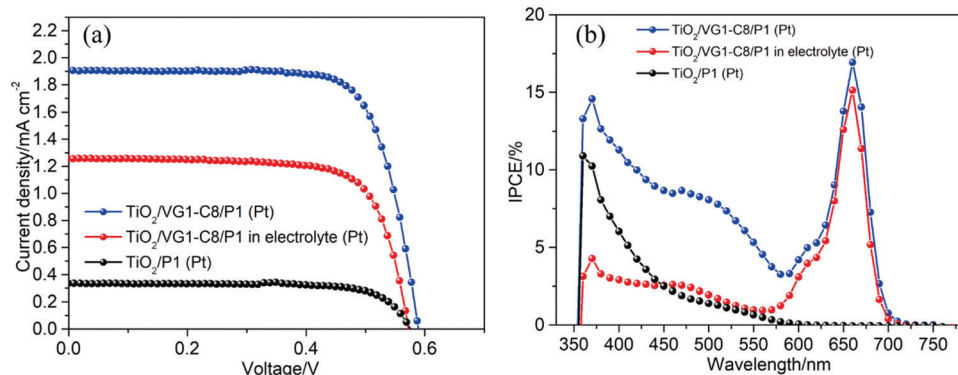


Fig. 10 (a) J - V curves of TiO_2 film sensitized VG1-C8 or P1 dye. (b) IPCE spectra of TiO_2 film sensitized VG1-C8 or P1 dye.



reaction is more plausible, because (1) FRET happens efficiently in dilute solution and on ZrO₂ film; (2) FRET is more long-range than electron transfer and may therefore happen faster than electron transfer between co-adsorbed dyes. However, it would not be surprising if a small portion of excited P1 dyes get electrons from VG1-C8 and then inject electrons to the TiO₂ CB.

Conclusion

In summary, the roles of NiO surface states in tandem dye-sensitized solar cells (t-DSSCs) have been assessed. The NiO surface states affect the charge transport processes in the tandem DSSCs and also participate in the IPCE response from both dyes on different photoelectrodes. By adjusting the photocurrent density of the TiO₂ photoanode in combination with the information from our previous work on NiO surface states, we can conclude that there are two pathways for electron transport through the NiO photocathode: (I) the electrons generated from the photoanode are compensated at the photocathode by the injected holes from excited dyes into NiO and (II) electrons generated at the photoanode go through the NiO film and directly reduce the redox couple (I⁻/I₃⁻), particularly when comparatively more electrons are generated at the photoanode than holes are being generated at the photocathode. Having a matching photocurrent from both photoelectrodes, meaning that an equal number of electrons and holes are generated at their respective electrodes, the sole presence of process I can render a high fill factor for t-DSSCs. If process II is involved to a large degree, the process will lower the FF factor of t-DSSCs due to unwanted resistance caused by the electron transport in or on the NiO film. Furthermore, due to the catalytic behavior of NiO surface states towards the electrolyte and the confirmed energy transfer between desorbed P1 and VG1-C8 dye on TiO₂, one can therefore explain the observed unexpected IPCE response from both dyes in a tandem device, despite their complementary optical absorption with little spectral overlap. This study provides new understandings to explain the strange IPCE phenomenon which has been existing in the research community for a long time.

Conflicts of interest

The authors declare no competing financial interest.

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References

- B. O'Regan and M. Grätzel, *Nature*, 1991, **353**, 737–740.
- A. Hagfeldt, G. Boschloo, L. Sun, L. Kloo and H. Pettersson, *Chem. Rev.*, 2010, **110**, 6595–6663.
- L. Zhang, X. Yang, W. Wang, G. G. Gurzadyan, J. Li, X. Li, J. An, Z. Yu, H. Wang, B. Cai, A. Hagfeldt and L. Sun, *ACS Energy Lett.*, 2019, **4**, 943–951.
- E. Benazzi, J. Mallows, G. H. Summers, F. A. Black and E. A. Gibson, *J. Mater. Chem. C*, 2019, **7**, 10409–10445.
- H. Tian, *Sustainable Energy Fuels*, 2019, **3**, 888–898.
- S. Sumikura, S. Mori, S. Shimizu, H. Usami and E. Suzuki, *J. Photochem. Photobiol., A*, 2008, **199**, 1–7.
- F. Odobel, L. Le Pleux, Y. Pellegrin and E. Blart, *Acc. Chem. Res.*, 2010, **43**, 1063–1071.
- B. Xu, L. Tian, A. S. Etman, J. Sun and H. Tian, *Nano Energy*, 2019, **55**, 59–64.
- D. Xiong and W. Chen, *Front. Optoelectron.*, 2012, **5**, 371–389.
- A. Nattestad, I. Perera and L. Spiccia, *J. Photochem. Photobiol., C*, 2016, **28**, 44–71.
- M. S. Prévot and K. Sivula, *J. Phys. Chem. C*, 2013, **117**, 17879–17893.
- F. Li, K. Fan, B. Xu, E. Gabrielsson, Q. Daniel, L. Li and L. Sun, *J. Am. Chem. Soc.*, 2015, **137**, 9153–9159.
- J. He, H. Lindström, A. Hagfeldt and S.-E. Lindquist, *J. Phys. Chem. B*, 1999, **103**, 8940–8943.
- J. He, H. Lindström, A. Hagfeldt and S.-E. Lindquist, *Sol. Energy Mater. Sol. Cells*, 2000, **62**, 265–273.
- M. Yu, G. Natu, Z. Ji and Y. Wu, *J. Phys. Chem. Lett.*, 2012, **3**, 1074–1078.
- M. Yu, T. I. Draskovic and Y. Wu, *Phys. Chem. Chem. Phys.*, 2014, **16**, 5026–5033.
- O. Langmar, E. Fazio, P. Schol, G. de la Torre, R. D. Costa, T. Torres and D. M. Guldi, *Angew. Chem., Int. Ed.*, 2019, **58**, 4056–4060.
- J. Qian, K.-J. Jiang, J.-H. Huang, Q.-S. Liu, L.-M. Yang and Y. Song, *Angew. Chem., Int. Ed.*, 2012, **51**, 10351–10354.
- Y. Hu, Z. Zheng, H. Jia, Y. Tang and L. Zhang, *J. Phys. Chem. C*, 2008, **112**, 13037–13042.
- S. Wrede and H. Tian, *Phys. Chem. Chem. Phys.*, 2020, **22**, 13850–13861.
- C. J. Wood, G. H. Summers and E. A. Gibson, *Chem. Commun.*, 2015, **51**, 3915–3918.
- A. Nattestad, A. J. Mozer, M. K. R. Fischer, Y. B. Cheng, A. Mishra, P. Bäuerle and U. Bach, *Nat. Mater.*, 2010, **9**, 31–35.
- L. D'Amario, L. J. Antila, B. Pettersson Rimgard, G. Boschloo and L. Hammarström, *J. Phys. Chem. Lett.*, 2015, **6**, 779–783.
- L. D'Amario, R. Jiang, U. B. Cappel, E. A. Gibson, G. Boschloo, H. Rensmo, L. Sun, L. Hammarström and H. Tian, *ACS Appl. Mater. Interfaces*, 2017, **9**, 33470–33477.
- L. D'Amario, G. Boschloo, A. Hagfeldt and L. Hammarström, *J. Phys. Chem. C*, 2014, **118**, 19556–19564.
- L. D'Amario, J. Föhlinger, G. Boschloo and L. Hammarström, *Chem. Sci.*, 2018, **9**, 223–230.
- N. T. Z. Potts, T. Sloboda, M. Wächtler, R. A. Wahyuono, V. D'Annibale, B. Dietzek, U. B. Cappel and E. A. Gibson, *J. Chem. Phys.*, 2020, **153**, 184704.



- 28 C. J. Wood, C. A. McGregor and E. A. Gibson, *ChemElectroChem*, 2016, **3**, 1827–1836.
- 29 M. Bonomo, D. Dini and A. G. Marrani, *Langmuir*, 2016, **32**, 11540–11550.
- 30 L. Tian, R. Tyburski, C. Wen, R. Sun, M. Abdellah, J. Huang, L. D'Amario, G. Boschloo, L. Hammarström and H. Tian, *J. Am. Chem. Soc.*, 2020, **142**, 18668–18678.
- 31 J. Park, C. Barolo, F. Sauvage, N. Barbero, C. Benzi, P. Quagliotto, S. Coluccia, D. Di Censo, M. Grätzel, M. K. Nazeeruddin and G. Viscardi, *Chem. Commun.*, 2012, **48**, 2782–2784.
- 32 P. Qin, H. Zhu, T. Edvinsson, G. Boschloo, A. Hagfeldt and L. Sun, *J. Am. Chem. Soc.*, 2008, **130**, 8570–8571.
- 33 F. Gong, H. Wang, X. Xu, G. Zhou and Z.-S. Wang, *J. Am. Chem. Soc.*, 2012, **134**, 10953–10958.
- 34 Z. Jin, M. Zhang, M. Wang, C. Feng and Z.-S. Wang, *Acc. Chem. Res.*, 2017, **50**, 895–904.
- 35 Z. Huang, G. Natu, Z. Ji, M. He, M. Yu and Y. Wu, *J. Phys. Chem. C*, 2012, **116**, 26239–26246.
- 36 J. Wei, Z. Shao, B. Pan, S. Chen, L. Hu and S. Dai, *Materials*, 2020, **13**, 2936.
- 37 P. Ho, S. Thogiti, L. Q. Bao, R. Cheruku, K.-S. Ahn and J. Hong Kim, *Solar Energy*, 2018, **161**, 9–16.
- 38 P. Ho, S. Thogiti, Y. H. Lee and J. H. Kim, *Sci. Rep.*, 2017, **7**, 2272.
- 39 Y. H. Lee, J. Y. Park, S. Thogiti, R. Cheruku and J. H. Kim, *Electron. Mater. Lett.*, 2016, **12**, 524–529.
- 40 H. Zhu, A. Hagfeldt and G. Boschloo, *J. Phys. Chem. C*, 2007, **111**, 17455–17458.
- 41 J. I. Basham, G. K. Mor and C. A. Grimes, *ACS Nano*, 2010, **4**, 1253–1258.
- 42 G. K. Mor, J. Basham, M. Paulose, S. Kim, O. K. Varghese, A. Vaish, S. Yoriya and C. A. Grimes, *Nano Lett.*, 2010, **10**, 2387–2394.

