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# Microelectrode-based transient amperometry of O<sub>2</sub> adsorption and desorption on a SrTiO<sub>3</sub> photocatalyst excited under water†

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Oxygen evolution at water–solid interfaces is a key reaction for sustainable energy production. Although some intermediate states have been detected in transient absorption spectroscopy, the O<sub>2</sub> evolution kinetics after the multi-step, four-electron oxidation of water remain unknown. In this study, transient amperometry with a microelectrode was applied to *operando* O<sub>2</sub> detection over Al-doped SrTiO<sub>3</sub> particles doubly loaded with RhCrO<sub>x</sub> and CoO<sub>y</sub> cocatalysts, an efficient photocatalyst for the overall water-splitting reaction. Electrochemical O<sub>2</sub> detection at intervals of 0.1 s unexpectedly indicated instantaneous O<sub>2</sub> adsorption and desorption in addition to steady, photocatalytic O<sub>2</sub> evolution on the photocatalyst modified under intense light irradiation. We hypothesized that electrons excited in the conduction band were transferred to O<sub>2</sub> in water through Ti cations neighboring an oxygen anion vacancy on the modified Al-doped SrTiO<sub>3</sub>. The negatively charged O<sub>2</sub> was then bound to the Ti cations. It was neutralized and released when shaded through electron back-transfer to the conduction band. The hypothesized mechanism for O<sub>2</sub> adsorption and desorption was compared with the photoinduced O<sub>2</sub> desorption known to occur on anion vacancies of TiO<sub>2</sub>(110). The microelectrode-based transient amperometry demonstrated in this paper will be applied to many other phenomena at liquid–solid interfaces.

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## Introduction

Oxygen evolution at water–solid interfaces is a key reaction for sustainable energy production.<sup>1–4</sup> In semiconductor photocatalysts and photoelectrodes, the creation and consumption kinetics of bandgap-excited charge carriers<sup>5–14</sup> or chemical intermediates<sup>15–20</sup> have been traced using transient absorption spectroscopy (TAS) with good time resolutions of femtoseconds to milliseconds. The kinetics observed by TAS for the initial and intermediate species must be compared with O<sub>2</sub> evolution

kinetics to verify or deny a proposed reaction mechanism. Nevertheless, it is still difficult to experimentally detect O<sub>2</sub> with a compatible time resolution.

Molecular oxygen released in water is currently accumulated in the gas phase and quantified with gas chromatography once or twice per hour. An improved time resolution on the scale of minutes could be achieved with advanced setups that shorten the gas-sampling intervals.<sup>21</sup> The authors have been developing a method using microelectrode-based transient amperometry to further speed up *operando* O<sub>2</sub> detection. Our simple idea is to shorten the physical distance from the place of reaction to the electrode for detection, as was conducted in scanning electrochemical microscopy (SECM) suitable for O<sub>2</sub> concentration mapping.<sup>22–26</sup> In our latest study,<sup>27</sup> we used diffusion simulations and electrochemical detection on a microelectrode to determine the absolute O<sub>2</sub> evolution rate with a time resolution of 0.1 s on a SrTiO<sub>3</sub> photocatalyst film immersed in a nitrogen-purged, aqueous KCl solution.

In the present study, the photocatalyst film was excited in an aerobic solution exposed to air. Surprisingly, molecular oxygen having been dissolved in the solution was adsorbed on the irradiated film, and it was released when shaded from the irradiation. We hypothesized that bandgap-excited electrons were injected into the O<sub>2</sub> in the solution. After receiving

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the electrons,  $O_2$  was negatively charged and bound to the Ti cations neighboring an oxygen anion vacancy on the surface of  $SrTiO_3$ .

## Experimental section

The photocatalyst was  $SrTiO_3$  particles doped with Al cations.<sup>28</sup>  $SrTiO_3$ ,  $Al_2O_3$ , and  $SrCl_2$  were mixed at a molar ratio of 100:2:10 and heated at 1423 K to produce Al-doped particles that were 200–500 nm in size. A  $RhCrO_x$  cocatalyst (Rh 0.1 wt% and Cr 0.1 wt%) was loaded onto the particles using impregnation from an aqueous solution of  $Na_3RhCl_6$  and  $Cr(NO_3)_3$ . A  $CoO_y$  cocatalyst (Co 0.1 wt%) was additionally loaded by photodeposition in an aqueous solution of  $CoCl_2$ . The  $RhCrO_x$  cocatalyst was loaded for assisting electron-driven  $H_2$  evolution,<sup>29</sup> whereas the  $CoO_y$  cocatalyst was for hole-driven  $O_2$  evolution.<sup>18,30</sup> This photocatalyst was active for the overall water-splitting reaction with an apparent quantum yield of 55%. The photocatalyst particles were suspended in water with  $SiO_2$  nanoparticles (1:2 weight ratio), dropped on a frosted glass plate, and dried at 323 K to be fixed. Photocatalyst-coated plates prepared in a similar manner had previously been placed in a panel reactor and tested for large-scale water splitting under natural sunlight.<sup>28</sup>

Details of the experimental setup are described in ref. 27 and its ESI. A millimeter-sized photocatalyst-coated glass plate was placed in a KCl aqueous solution ( $0.1 \text{ mol l}^{-1}$ , pH = 7) that was exposed to air in this study. A platinum wire with a  $10 \mu\text{m}$  radius was coated with glass to expose a metal section as the working electrode for  $O_2$  detection. The electrode was immersed in the solution perpendicular to the photocatalyst film which was electrically isolated in the solution. The distance between the electrode and film was controlled in the 100–200  $\mu\text{m}$  range with  $0.1 \mu\text{m}$  precision. Molecular oxygen in the solution was electrochemically detected on the electrode with a time resolution of 0.1 s, which was biased at  $-0.5 \text{ V}$  relative to a Ag wire as the counter electrode. A four-electron reduction reaction,  $O_2 + 2H_2O + 4e^- \rightarrow 4OH^-$  and/or  $O_2 + 4H^+ + 4e^- \rightarrow 2H_2O$ ,<sup>31</sup> produced an electrode current on the order of nano-amperes that was proportional to the  $O_2$  concentration. The proportional relation of  $O_2$  concentration and electrode current was checked and confirmed.<sup>27</sup> The Ag wire served as a quasi Ag/AgCl reference electrode in the KCl solution. On the other hand, the oxidative reaction on the wire,  $Ag + h^+ \rightarrow Ag^+$ , as the counter electrode caused silver contamination in the solution to a limited extent. A freshly polished microelectrode was used in each set of measurements to minimize silver contamination on the microelectrode apex.

## Results and discussion

The immersed film was side lit with ultraviolet (UV) light (center wavelength: 280 nm) passing through the solution, as illustrated in the table of contents entry. The light intensity was tuned to  $3 \text{ W m}^{-2}$  ( $4 \times 10^{18} \text{ photons m}^{-2} \text{ s}^{-1}$ ) to probe light-triggered  $O_2$  concentration changes without producing  $O_2$  and  $H_2$  bubbles on the irradiated film. Fig. 1(A) presents the

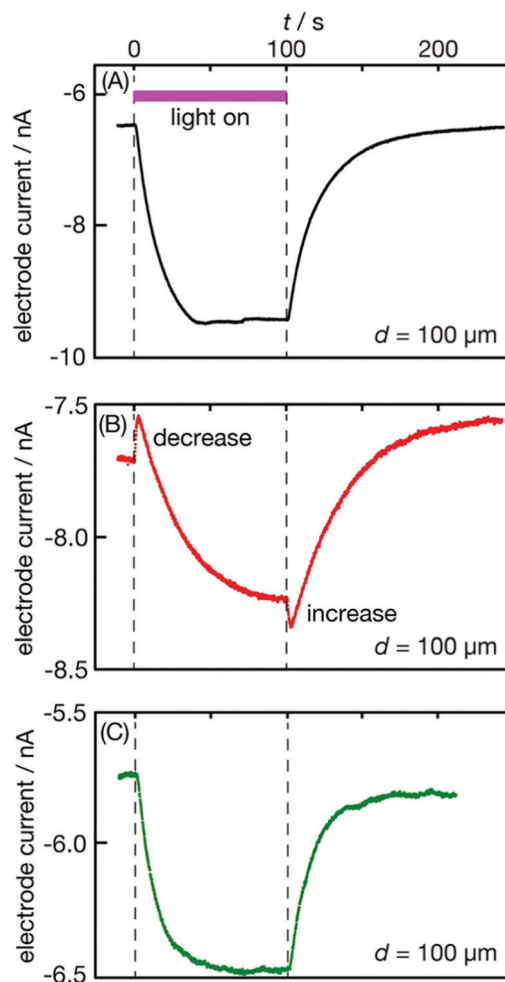


Fig. 1 Microelectrode current in response to UV light irradiation. Irradiation of a photocatalyst film with light for probing (intensity:  $3 \text{ W m}^{-2}$ ) at  $t = 0$ – $100.5 \text{ s}$  in a KCl solution resulted in response (A). The film was then irradiated with intense light to modify the photocatalyst ( $30 \text{ W m}^{-2}$ ) for 4 h in the solution. Immediate irradiation of the modified film with light for probing ( $3 \text{ W m}^{-2}$ ) at  $t = 0$ – $100.2 \text{ s}$  resulted in response (B). The instantaneous current decrease and increase are marked in the panel. The film was removed from the solution and dried in air for 9 days. The dried film showed response (C) when immersed in the solution and irradiated with light for probing ( $3 \text{ W m}^{-2}$ ) at  $t = 0$ – $101.1 \text{ s}$ . The electrode–film distance  $d$  was set at  $100 \mu\text{m}$ .

electrode current  $I$  that responded to light irradiation at an electrode–film distance  $d = 100 \mu\text{m}$ . The negative sign of the current indicates that electrons are transferred from the electrode to the solution. Molecular oxygen in the solution produced a background current of  $-6.5 \text{ nA}$ . When the probe light was turned on at time zero ( $t = 0$ ), the electrode current gradually increased to  $-9.5 \text{ nA}$ ; when the light stopped at  $100.5 \text{ s}$ , the current gradually decreased back to the background.

We had observed gradual current increases and decreases in response to UV light irradiation on the same photocatalyst,<sup>27</sup> where the whole setup was purged with  $N_2$  to remove background  $O_2$  in the solution. The current changes observed in the  $N_2$  atmosphere were quantitatively interpreted with photocatalytic  $O_2$  evolution on the irradiated film followed by  $O_2$



diffusion into the bulk solution. The electrode current response shown in Fig. 1(A) increased and then decreased in a similar manner to what was reported in ref. 27 Hence, the response in Fig. 1(A) was ascribed to photocatalytic O<sub>2</sub> evolution on the irradiated film.

The film examined in Fig. 1(A) was then irradiated with UV light at 10 times the intensity (280 nm, 30 W m<sup>-2</sup>) for 4 h in the solution. The film was photocatalytically active enough to produce O<sub>2</sub> and H<sub>2</sub> bubbles during this intense irradiation. The electrode current exceeded the quantification limit, 10.5 nA. When the intense irradiation was suspended, the film was immediately probed with weak light (3 W m<sup>-2</sup>). Response (B) obtained in this way was qualitatively different from response (A). The electrode current was stable at -7.7 nA, indicating that O<sub>2</sub> was dissolved in the air-exposed solution prior to irradiation. When the probe light was switched on, the current instantaneously decreased from -7.7 to -7.5 nA. It then gradually increased, indicating photocatalytic O<sub>2</sub> production. When the probe light was stopped at  $t = 100.2$  s, the electrode current instantaneously increased from -8.3 to -8.4 nA and gradually reduced back to the background level. We ascribed the instantaneous current increase coupled with the decrease to O<sub>2</sub> adsorption and desorption on the film triggered by light irradiation; background O<sub>2</sub> in the solution was adsorbed on the film when irradiated and was desorbed when shaded.

Pretreatment with intense light irradiation (30 W m<sup>-2</sup> for 4 h) was required to modify the photocatalyst feasible for probe-light-triggered O<sub>2</sub> adsorption and desorption. Immersing the photocatalyst film in the solution for 4 h without intense light irradiation provided no instantaneous current response, as shown in Fig. S1 in the ESI.† Intense irradiation on the microelectrode induced no current response in the absence of the film (Fig. S2, ESI†). In a K<sub>2</sub>SO<sub>4</sub> solution (0.1 mol l<sup>-1</sup>), the film was modified under intense light to present similar instantaneous current responses (Fig. S3, ESI†) being insensitive to the chemical identity of electrolytes. Oxidation of Cl<sup>-</sup> anions, if any on the UV-irradiated SrTiO<sub>3</sub> photocatalyst, did not matter to make the instantaneous current responses.

These experimental results supported our interpretation that probe-light-triggered O<sub>2</sub> adsorption and desorption occurred on the photocatalyst particles modified during the pretreatment.

After the film displayed response (B), it was removed from the solution and dried in air at room temperature for 9 days. This film was then immersed in the KCl solution and examined with probe light (3 W m<sup>-2</sup>) again. The observed response (C) displayed a gradual current increase and decrease, which were attributed to photocatalytic O<sub>2</sub> evolution; no instantaneous response could be assigned to O<sub>2</sub> adsorption and desorption. The modified photocatalyst particles returned to their original state when dried in air.

In the descriptions above, we qualitatively characterized the electrode current responses according to the presence or absence of instantaneous responses triggered by probe-light irradiation. We then quantitatively interpreted response (B) in Fig. 1, using a numerical simulation including

instantaneous O<sub>2</sub> adsorption, photocatalytic O<sub>2</sub> evolution, instantaneous O<sub>2</sub> desorption, and O<sub>2</sub> diffusion in the solution. The simulation space defined in cylindrical coordinates,  $r$  and  $z$ , is illustrated in Fig. 2(A). The oxygen diffusion driven by a concentration gradient is described by Fick's law:

$$\frac{\partial C}{\partial t} = D \left( \frac{\partial^2 C}{\partial r^2} + \frac{1}{r} \frac{\partial C}{\partial r} + \frac{\partial^2 C}{\partial z^2} \right)$$

where  $C$  is concentration, and  $D$  is the diffusion coefficient of O<sub>2</sub> ( $D = 2.5 \times 10^{-5}$  cm<sup>2</sup> s<sup>-1</sup> at 298 K).<sup>32</sup> The details of the finite-element scheme, which integrates the equation to predict the electrode current, are described with Fig. S4 in ESI.† To fit the experimentally observed response shown in Fig. 1(B), we assumed probe-light-triggered O<sub>2</sub> adsorption at  $t = 0 - \tau$ , photocatalytic O<sub>2</sub> evolution at  $t = \tau - 100.2$  s, and O<sub>2</sub> desorption at  $t = 100.2$  s - (100.2 s +  $\tau$ ), as depicted in Fig. 2(B). Positive and negative O<sub>2</sub> release rates in the unit of  $\mu\text{mol m}^{-2} \text{s}^{-1}$  were used to represent O<sub>2</sub> release and suction on the film, respectively. The time width  $\tau$  was not necessarily identical for the adsorption and desorption phenomena. However, a common  $\tau$  value was assumed to minimize the number of fitting parameters.

The O<sub>2</sub> release rate depicted in Fig. 2(B) represented a net release rate. When light-triggered O<sub>2</sub> adsorption or desorption occurred with photocatalytic O<sub>2</sub> evolution in parallel, the net release rate was evaluated by fitting simulated current responses to the experimental result.

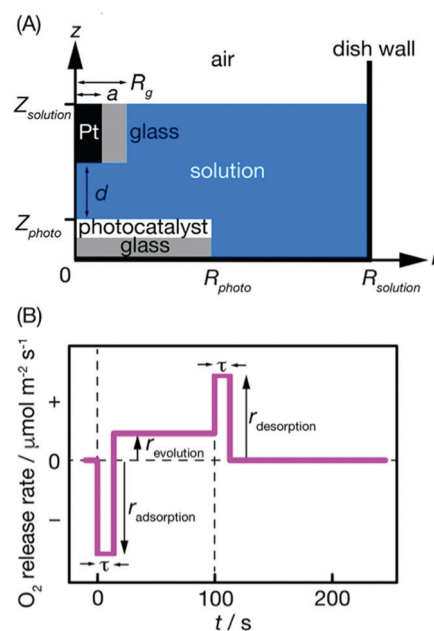


Fig. 2 Finite-element simulation of electrode current. The simulation space, which has cylindrical symmetry around the  $z$ -axis, is illustrated in (A). Platinum electrode section radius,  $a$ : 10  $\mu\text{m}$ ; glass wrap outer radius,  $R_g$ : 20  $\mu\text{m}$ ; electrode-film distance,  $d$ : 100  $\mu\text{m}$ ; radius of the irradiated photocatalyst film,  $R_{\text{photo}}$ : fitted to be 800  $\mu\text{m}$  in the simulation (see text); solution dish radius,  $R_{\text{solution}}$ : 13 mm; solution thickness,  $Z_{\text{solution}}$ : 10 mm; glass plate thickness,  $Z_{\text{photo}}$ : 1 mm. The solid line in (B) depicts the assumed O<sub>2</sub> release rate on the film.



The simulated current was fitted to the experimental result in two steps. In the first step,  $R_{\text{photo}}$  and  $r_{\text{evolution}}$  were optimized to follow the gradual current increase during probe-light irradiation, while  $\tau$  was set to zero. The other quantities ( $a$ ,  $R_{\text{g}}$ ,  $d$ ,  $R_{\text{solution}}$ ,  $Z_{\text{solution}}$ , and  $Z_{\text{photo}}$ ) were fixed at experimentally determined values, as described in the caption of Fig. 2. The radius of the irradiated photocatalyst film  $R_{\text{photo}}$  was difficult to quantitatively determine since the light-spot boundary was oval and graded on the film. Thus,  $R_{\text{photo}}$  and  $r_{\text{evolution}}$  were optimized to fit the experimentally observed response. The obtained  $R_{\text{photo}}$  was  $800 \mu\text{m}$ , while  $r_{\text{evolution}}$  was  $0.085 \mu\text{mol m}^{-2} \text{s}^{-1}$ . In the second step,  $r_{\text{adsorption}}$  and  $\tau$  were fitted to the observed response for  $t = 0-10$  s. Fitting to the response in  $t = 0-10$  s was long enough to evaluate the impulsive  $\text{O}_2$  suction having occurred at  $t = 0 - \tau$ , since the time delay for diffusion across the electrode-film gap was characterized by  $d^2/D = 4$  s.

When irradiation was stopped,  $r_{\text{desorption}}$  and  $\tau$  were optimized to follow the observed response at  $t = 100-110$  s. The current response simulated in this way was compared with the observed response in Fig. 3, in which the background current ( $-7.7$  nA) was subtracted to deduce the current change induced by probe-light irradiation. The simulated response reproduced the observed current in the three time windows,  $t = 0-250$ ,  $0-10$ , and  $95-110$  s, and the optimized parameters were  $r_{\text{adsorption}} = -0.35 \mu\text{mol m}^{-2} \text{s}^{-1}$ ,  $r_{\text{desorption}} = 0.21 \mu\text{mol m}^{-2} \text{s}^{-1}$ , and  $\tau = 2.0$  s. Two more response curves were experimentally observed on the same film upon changing  $d$  to 150 and  $200 \mu\text{m}$  (Fig. 4). The two curves were fitted to validate the simulation scheme and the optimized parameters.

A negative release rate of  $-0.35 \mu\text{mol m}^{-2} \text{s}^{-1}$  was evaluated in the first 2.0 s of probe-light irradiation. The integrated quantity of adsorbed  $\text{O}_2$ , which was calculated as the product of  $r_{\text{adsorption}}$  and  $\tau$ , had a value of  $4.2 \times 10^{17}$  molecules  $\text{m}^{-2}$ . The integrated quantity corresponded to 6% of the surface unit cell density of crystalline  $\text{SrTiO}_3$ ,  $6.6 \times 10^{18} \text{m}^{-2}$  on its (100) truncation.<sup>33</sup> The integrated quantity suggests  $\text{O}_2$  adsorption on  $\text{SrTiO}_3$  rather than on the  $\text{RhCrO}_x$  and/or  $\text{CoO}_y$  cocatalysts loaded by 0.1 wt%. Partial contribution of the cocatalysts cannot be excluded, though. Since the film was composed of particulate materials, the unit cell density exposed to the solution should be larger than this number. Hence, the integrated quantity of 6% relative to the unit cell density provides the maximum estimate.

The integrated quantity of desorbed  $\text{O}_2$ , which was calculated as the product of  $r_{\text{desorption}}$  and  $\tau$ , had a value of  $2.5 \times 10^{17}$  molecules  $\text{m}^{-2}$ . The desorbed quantity was 40% smaller than the adsorbed quantity. The decremental difference was ascribed to  $\text{O}_2$  desorption during probe-light irradiation or to the conversion of the adsorbed  $\text{O}_2$  to resident species on the shaded film.

Finally, we considered the mechanism through which  $\text{O}_2$  in the solution was adsorbed on the Al-doped  $\text{SrTiO}_3$  particles. Several peroxotitanate complexes having Ti metal centers coordinated by  $\text{O}_2^{2-}$  ligands were previously reported.<sup>34</sup> We therefore hypothesize that negatively charged  $\text{O}_2$  species, possibly  $\text{O}_2^{2-}$ , were bound to Ti cations on the surface of the  $\text{SrTiO}_3$  during probe-light irradiation. Bandgap excitation by probe

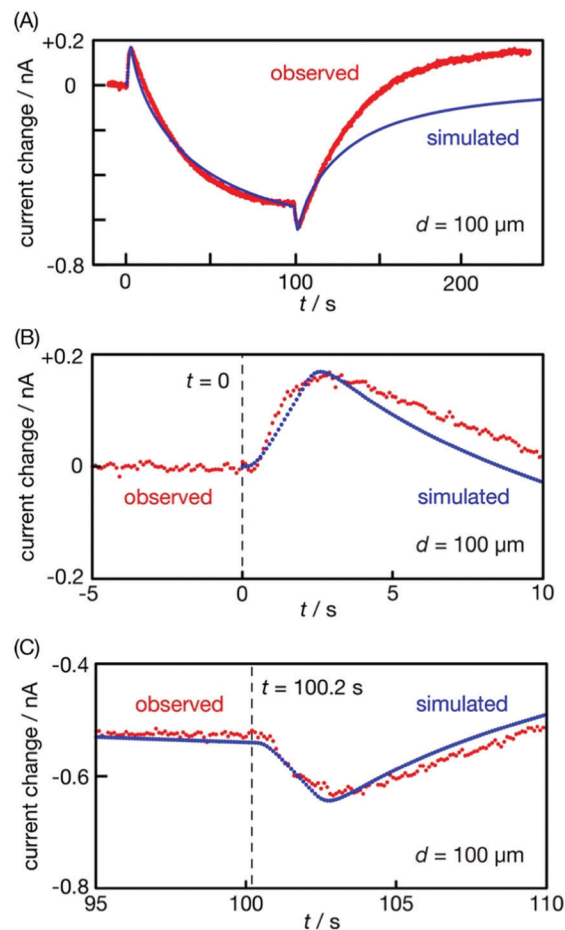


Fig. 3 Electrode current response shown in Fig. 1(B) fitted with the finite-element simulation of  $\text{O}_2$  adsorption, evolution, and desorption on the photocatalyst film. The experimentally observed (red) and simulated (blue) current changes are shown in time windows of (A) 0–250, (B) 0–10, and (C) 95–110 s. Electrode–film distance  $d = 100 \mu\text{m}$ . UV light ( $3 \text{ W m}^{-2}$ ) was irradiated for probing on the photocatalyst film at  $t = 0-100.2$  s. Five fitting parameters were optimized:  $R_{\text{photo}} = 800 \mu\text{m}$ ,  $r_{\text{evolution}} = 0.085 \mu\text{mol m}^{-2} \text{s}^{-1}$ ,  $r_{\text{adsorption}} = -0.35 \mu\text{mol m}^{-2} \text{s}^{-1}$ ,  $r_{\text{desorption}} = 0.21 \mu\text{mol m}^{-2} \text{s}^{-1}$ , and  $\tau = 2.0$  s.

light provided two electrons, which were transferred to an impinging  $\text{O}_2$  molecule. By shading the particles, the electrons were back-transferred to the photocatalyst particle, resulting in the release of the neutralized  $\text{O}_2$  into the solution.

Titanium cations exposed on pristine particles were unable to capture  $\text{O}_2$  even under the probe light, as was evidenced by the absence of the instantaneous current response in Fig. 1(A). The surface of the particles was modified during the pretreatment under intense light irradiation ( $30 \text{ W m}^{-2}$ ) to enable the capture of negatively charged  $\text{O}_2$ . The modified surface returned to its original state when the film was dried in air at room temperature. The reversible state changes suggested that a limited number of oxygen anions were removed during the pretreatment and resultant anion vacancies served as active site for  $\text{O}_2$  adsorption. The vacancies were healed on the dried film. If metal (Sr, Ti, Al, Cr, Rh or Co) cations were removed under irradiation and induced the light-triggered adsorption, then they could not be restored when the surface was dried in air.





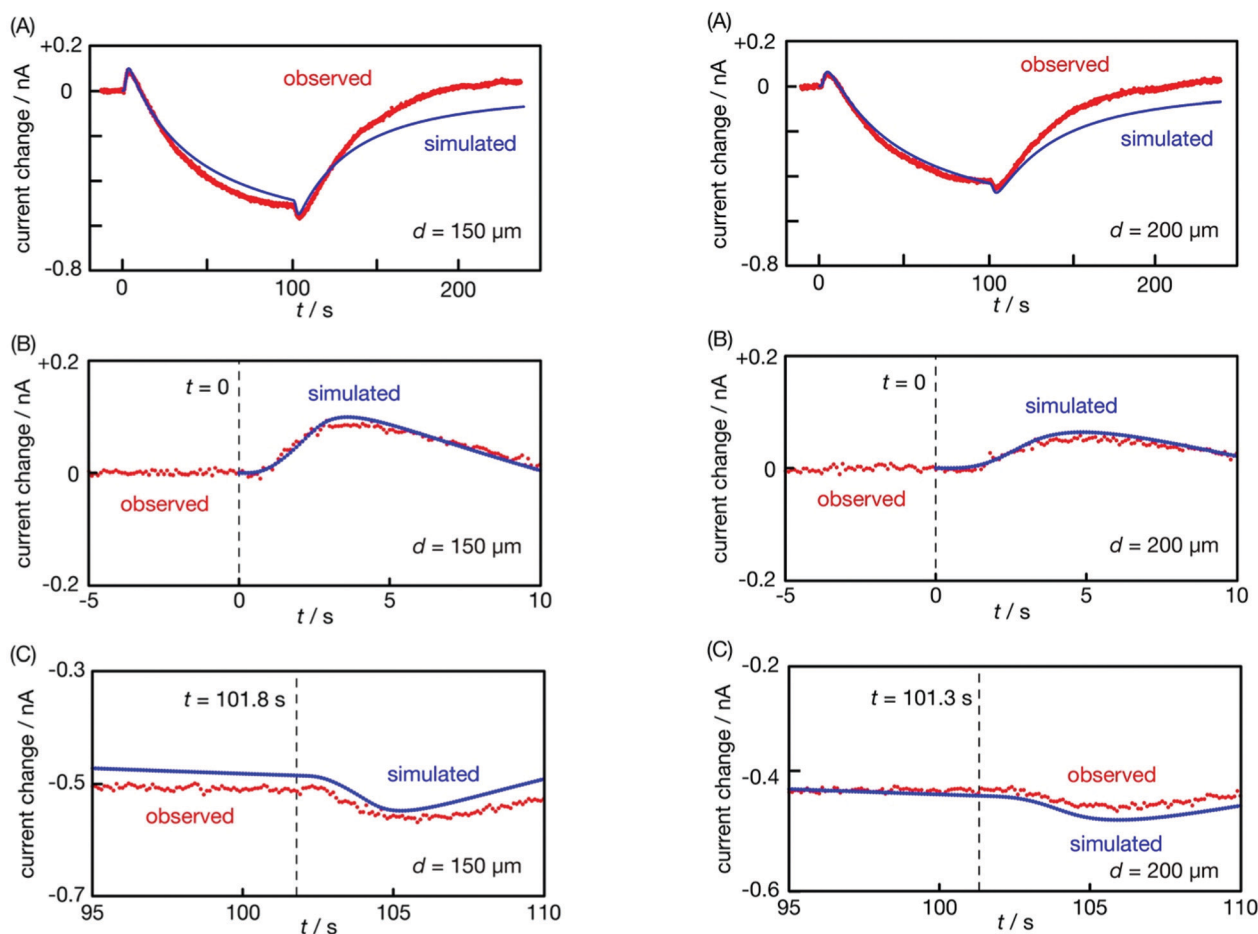


Fig. 4 Current changes at  $d =$  (left panel) 150 and (right panel) 200  $\mu\text{m}$ . The experimentally observed (red) and simulated (blue) current changes are shown in time windows of (A) 0–250, (B) 0–10, and (C) 95–110 s. UV light for probing ( $3\text{ W m}^{-2}$ ) was irradiated on the film at  $t =$  (left panel) 0–101.8 and (right panel)  $t =$  0–101.3 s. The five fitting parameters ( $R_{\text{photo}}$ ,  $f_{\text{evolution}}$ ,  $f_{\text{adsorption}}$ ,  $f_{\text{desorption}}$ , and  $\tau$ ) were fixed at values optimized in the simulation at  $d = 100\ \mu\text{m}$ .

An analogous mechanism for  $\text{O}_2$  adsorption and desorption is known to occur on  $\text{TiO}_2(110)$  wafers placed in vacuum. Experimental<sup>35,36</sup> and theoretical<sup>37</sup> studies have concluded that  $\text{O}_2$  was adsorbed on  $\text{Ti}^{3+}$  cations that neighbored an oxygen anion vacancy when the vacancy was thermally created on the wafer surface. The adsorbed  $\text{O}_2$  receives two excess electrons that were originally localized on the  $\text{Ti}^{3+}$  cations. When the wafer is bandgap excited, two holes attached to the negatively charged  $\text{O}_2$  and the neutralized  $\text{O}_2$  escaped as gas. Electron transfer and back-transfer switched the  $\text{O}_2$  adsorption and desorption on the anion vacancy.

We further hypothesized how oxygen anions were removed from Al-doped  $\text{SrTiO}_3$  particles during the pretreatment. Upon capture of two holes, an oxygen anion is neutralized and leave the surface. This process is known as photocorrosion;<sup>38</sup> however,  $\text{SrTiO}_3$  is recognized as a stable semiconductor against photocorrosion. The neutralization of an oxygen anion leading to vacancy creation thermodynamically and kinetically competes with water oxidation, when a metal oxide particle is excited under water. The electrochemical potential for this phenomenon, *i.e.*, corrosion potential, is known to be 1.42 eV

*vs.* NHE on  $\text{TiO}_2$  in water having a pH of 0. On the other hand, the potential for water oxidation is 1.23 eV at pH 0. When we assume an identical corrosion potential on the surface of Al-doped  $\text{SrTiO}_3$ , photoexcited holes are thermodynamically transferred to water and not to surface oxygen anions.

A finite number of holes are still feasible for neutralizing oxygen anions when water oxidation is kinetically limited. The Al-doped  $\text{SrTiO}_3$  photocatalyst examined in this study was doubly loaded with the  $\text{CrRhO}_x$  cocatalyst for electron-driven  $\text{H}_2$  production<sup>29</sup> and the  $\text{CoO}_y$  cocatalyst for hole-driven  $\text{O}_2$  production.<sup>28</sup> A finite number of holes could have been present in Al-doped  $\text{SrTiO}_3$  particles even modified with the  $\text{CoO}_y$  cocatalyst for water oxidation.

Thus, the authors hypothesized that a small number of oxygen anions were neutralized and removed during the pretreatment under intense light irradiation. Indeed, a previous study<sup>39</sup> reported photoetching of  $\text{TiO}_2$  electrodes which is known to be thermodynamically stable oxide against photocorrosion. The self-oxidation of the cocatalysts, if any during the pretreatment, may have created oxygen vacancies on the cocatalysts and contributed to the probe-light induced  $\text{O}_2$  adsorption and desorption.



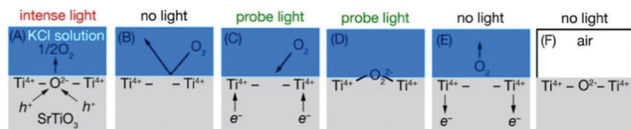


Fig. 5 A possible mechanism for light-triggered  $\text{O}_2$  adsorption on the Al-doped  $\text{SrTiO}_3$  photocatalyst particles. (A) A surface oxygen anion neutralized and removed during intense light irradiation ( $30 \text{ W m}^{-2}$ ). (B) No  $\text{O}_2$  adsorption to the vacancy in the electronic ground state. (C)  $\text{O}_2$  adsorption to the vacancy, assisted by electrons excited under probe-light irradiation ( $3 \text{ W m}^{-2}$ ). (D) Adsorption of a negatively charged  $\text{O}_2$  on the vacancy. (E)  $\text{O}_2$  desorption under shaded conditions, with electrons leaving. (F) The anion vacancy slowly healed in air at room temperature.

The hypothesized mechanism is depicted in Fig. 5. In panel (A), a small number (6% or less relative to surface lattice density) of surface oxygen anions of Al-doped  $\text{SrTiO}_3$  are neutralized and removed during the pretreatment. The large number of holes excited under intense light irradiation drive this step. Excited electrons corresponding to the holes that neutralize the anions are consumed in the  $\text{H}_2$  evolution reaction,  $2\text{H}_2\text{O} + 2e^- \rightarrow \text{H}_2 + 2\text{OH}^-$ . The  $\text{OH}^-$  anions left in the reaction can be released in the solution. The ionic charges in the particle are balanced without need of creating  $\text{Ti}^{3+}$  cations, as a result. (B)  $\text{O}_2$  cannot be adsorbed on the anion vacancy when shaded. (C) Under probe-light irradiation, electrons excited in the particle are transferred to an impinging  $\text{O}_2$  through Ti cations neighboring the vacancy. (D) The negatively charged  $\text{O}_2$  is bound to the vacancy. (E) When the probe light is turned off, the excited electrons leave the adsorbed  $\text{O}_2$ , and the neutralized  $\text{O}_2$  is released into the solution. The electrons leaving the adsorbed  $\text{O}_2$  are consumed in the  $\text{H}_2$  evolution reaction. (F) The anion vacancies were slowly healed after the film was dried in air.

$\text{O}_2$  desorption leaving an oxygen anion vacancy, which was hypothesized in Fig. 5(E), is identical to the  $\text{O}_2$  release step in the lattice oxygen evolution reaction (LOER), e.g., Fig. 4 of ref. 3 LOER has been recently proposed as an electrochemical cycle of the  $\text{O}_2$  evolution reaction on perovskite-structured metal oxide electrodes; it is different from the proton-coupled electron transfer<sup>4</sup> that was previously assumed as the mechanism.

To investigate our hypothesis, we compared the photocatalyst films before and after the pretreatment using X-ray photoelectron spectroscopy (Fig. S5, ESI<sup>†</sup>) and optical absorption (Fig. S6, ESI<sup>†</sup>). The results show no sign of reduced Ti cations. This is consistent with the proposed scheme (Fig. 5), which involves Ti cations only in the 4+ valence state.

The  $\text{SrTiO}_3$  particles were doped with Al cations in this study to enhance water-splitting efficiency by limiting electron-hole recombination. Transient absorption studies<sup>40,41</sup> showed recombination rate in  $\text{SrTiO}_3$  particles decreased by Na doping, whereas recombination rate was not yet examined on Al-doped  $\text{SrTiO}_3$ . It is possible that the Al cations embedded in  $\text{SrTiO}_3$  lattice played a positive or negative role in removing oxygen anions and in the instantaneous  $\text{O}_2$  responses. This possibility should be examined in the future.

The light-triggered  $\text{O}_2$  adsorption and desorption found in this study open an additional path for electron-hole recombination. Two excited electrons can be located in an  $\text{O}_2$  adsorbed on the vacancy. These electrons recombine with two holes, thus releasing the adsorbed  $\text{O}_2$  under steady light irradiation. Two electrons and holes are consumed to catch and release an  $\text{O}_2$  molecule in the solution without splitting water.

In addition, the hypothesized hole-induced vacancy creation is a side-reaction that is problematic for photocatalyst durability. Durability of the Al-doped  $\text{SrTiO}_3$  photocatalyst was enhanced over a span of 1300 h under simulated sunlight irradiation by loading the  $\text{CoO}_y$  cocatalyst.<sup>30</sup> The limited durability in the absence of  $\text{CoO}_y$  was interpreted with hole-induced degradation of the  $\text{RhCrO}_x$  cocatalyst. The hole-induced vacancy creation hypothesized in this study may have been an additional reason of the limited durability.

A semiconductor photocatalyst that efficiently drives the water-splitting reaction is potentially modified by the hole-induced creation of anion vacancies since electron-hole recombination is limited to increased quantum yields. Note that the light intensity for vacancy creation was  $30 \text{ W m}^{-2}$  near UV intensity in natural sun light ( $40 \text{ W m}^{-2}$  for wavelengths shorter than 400 nm). When this occurs on a photocatalyst under development, the tuning of cocatalysts to accelerate water-induced, hole-consuming reactions can be a beneficial way to optimize the number of holes present in the semiconductor particles under excitation light of a given intensity.

## Conclusion

In this study, we found that the  $\text{O}_2$  adsorption and desorption on the Al-doped  $\text{SrTiO}_3$  photocatalyst particles doubly loaded with  $\text{RhCrO}_x$  and  $\text{CoO}_y$  cocatalysts were triggered by photoexcitation in water. Electrochemical detection on the microelectrode combined with diffusion simulations allowed us to quantitatively determine the rates of  $\text{O}_2$  adsorption,  $\text{O}_2$  desorption, and photocatalytic  $\text{O}_2$  evolution. The time resolution for  $\text{O}_2$  detection, 0.1 s, was critical for recognizing the instantaneous  $\text{O}_2$  adsorption and desorption that occurred during steady  $\text{O}_2$  evolution. The transient amperometry using a microelectrode can be applied to many other phenomena at liquid–solid interfaces.

The light-triggered  $\text{O}_2$  adsorption and desorption were interpreted as the excited-electron transfer to and back-transfer from the adsorbed  $\text{O}_2$ . Upon receiving electrons,  $\text{O}_2$  in the solution was negatively charged and could bind to the surface  $\text{Ti}^{4+}$  cations, while shaded particles could neutralize and release  $\text{O}_2$  by back-transferring the electrons. Intense light irradiation ( $30 \text{ W m}^{-2}$ ) was required to modify the particle to enable  $\text{O}_2$  adsorption. We hypothesized that a small number of lattice oxygen anions were neutralized by receiving holes to leave the surface under intense irradiation. Anion vacancies created in this way served as  $\text{O}_2$  adsorption sites. Future studies that examine the role of the Al cations,  $\text{RhCrO}_x$  cocatalyst, and  $\text{CoO}_y$  cocatalyst are needed to verify the hypothesized scheme.



## Conflicts of interest

There are no conflicts to declare.

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