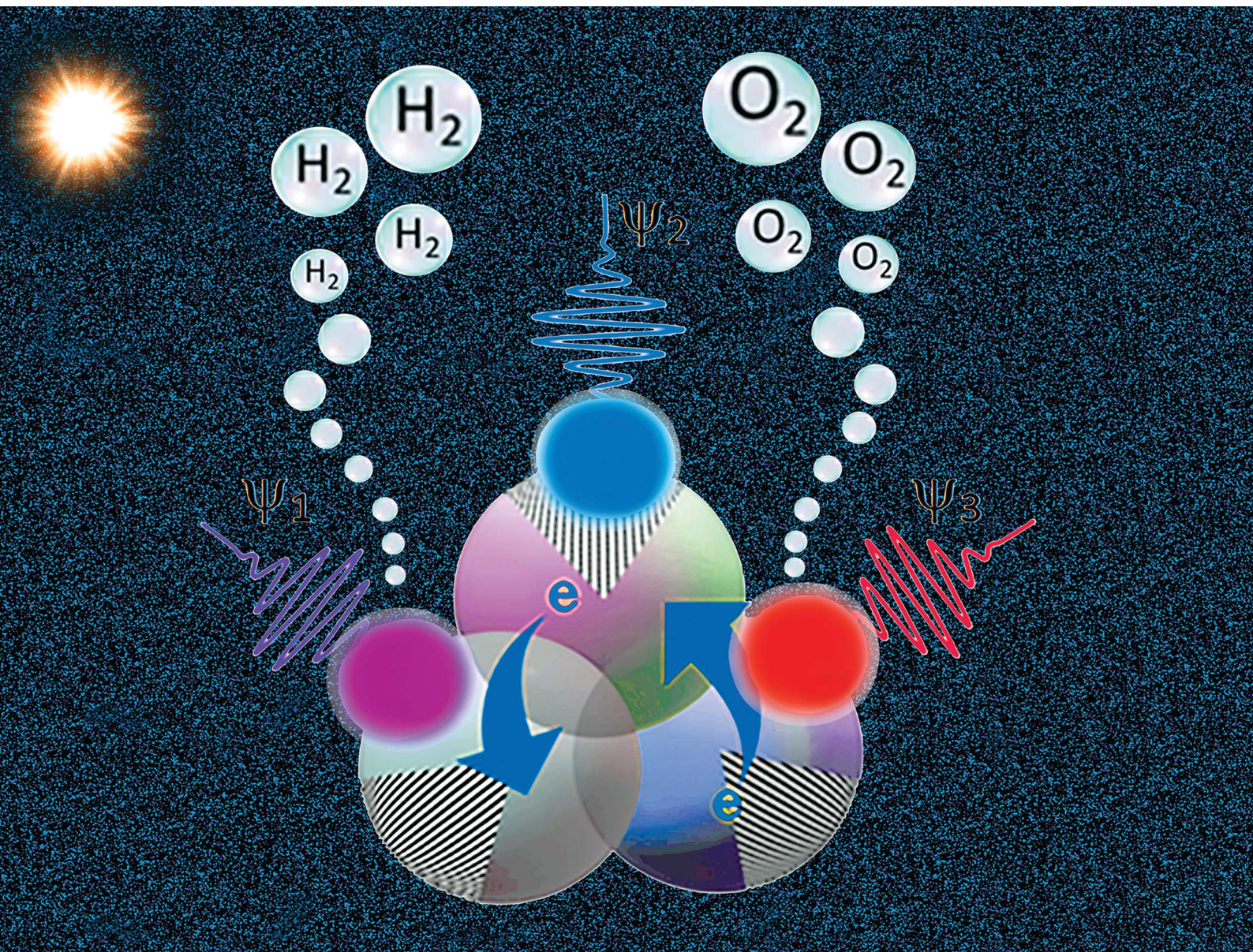


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## A photoanode with plasmonic nanoparticles of earth abundant bismuth for photoelectrochemical reactions†

Palyam Subramanyam,<sup>a</sup> Melepurath Deepa,<sup>a</sup> Sai Santosh Kumar Raavi,<sup>ID a</sup> Hiroaki Misawa,<sup>bc</sup> Vasudevanpillai Biju<sup>ID \*c</sup> and Challapalli Subrahmanyam<sup>ID \*a</sup>

A wide range of technologies has been developed for producing hydrogen economically and in greener ways. Photoelectrochemical water splitting using photoelectrodes submerged in a bath electrolyte forms a major route of hydrogen evolution. The efficacy of water splitting is improved by sensitizing metal oxide photoelectrodes with narrow bandgap semiconductors that efficiently absorb sunlight and generate and transport charge carriers. Here we show that the efficiencies of photocurrent generation and photoelectrochemical hydrogen evolution by the binary  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  anode increase by an order of magnitude upon the incorporation of the earth-abundant plasmonic bismuth nanoparticles into it. The ternary electrode  $\text{TiO}_2/\text{Bi}$  nanoparticle/ $\text{Sb}_2\text{S}_3$  illuminated with sunlight provides us with a photocurrent density as high as  $4.21 \text{ mA cm}^{-2}$  at 1.23 V, which is fourfold greater than that of the binary electrode and tenfold greater than that of pristine  $\text{TiO}_2$ . By using bismuth nanoparticles, we estimate the incident photon to current conversion efficiency at 31% and solar power conversion efficiency at 3.85%. Here the overall impact of bismuth nanoparticles is attributed to increases in the open-circuit voltage (860 mV), which is by expediting the transfer of photogenerated electrons from  $\text{Sb}_2\text{S}_3$  nanoparticles to the  $\text{TiO}_2$  electrode, and short-circuit current ( $9.54 \text{ mA cm}^{-2}$ ), which is by the plasmonic nearfield effect. By combining the cost-effective plasmonic bismuth nanoparticles with the narrow bandgap  $\text{Sb}_2\text{S}_3$  on the  $\text{TiO}_2$  electrode, we develop a stable, ternary photoanode and accomplish high-efficiency photocurrent generation and hydrogen evolution.

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Hydrogen is the most promising fuel for fulfilling the increasing global energy demand and cossetting the environmental costs of fossil fuels and greenhouse gases.<sup>1–3</sup> Photoelectrochemical (PEC) water splitting for hydrogen production is environmentally benign.<sup>4–7</sup> Even though  $\text{TiO}_2$  is a promising and the most investigated n-type semiconductor photoelectrode for PEC water splitting, its efficiency to produce hydrogen is limited by the large bandgap (3.2 eV) or the absorption of UV radiation.<sup>8–11</sup> Therefore,  $\text{TiO}_2$  is modified with narrow bandgap semiconductors such as CdS,  $\text{BiVO}_4$  and  $\text{Bi}_2\text{S}_3$ , which is for extending the absorption edge to the visible and near infrared regions.<sup>12–17</sup> For example, an  $\text{FTO}/\text{TiO}_2/\text{BiVO}_4$  heterojunction electrode provides higher PEC performance ( $4.11 \text{ mA cm}^{-2}$  at 1.23 V) over  $\text{FTO}/\text{TiO}_2$  or  $\text{FTO}/\text{BiVO}_4$ .<sup>18</sup> Similarly, heterojunction

photoelectrodes composed of  $\text{TiO}_2$  nanorods and  $\text{Bi}_2\text{S}_3$  nanowires provide higher photocurrent density ( $2.4 \text{ mA cm}^{-2}$  at 0.7 V vs. RHE) than pristine  $\text{TiO}_2$ .<sup>19</sup>

Along with the natural abundance of  $\text{Sb}_2\text{S}_3$  and the large optical absorption coefficient, the high stability and narrow bandgap (1.7–1.8 eV) make it a promising co-sensitizer of  $\text{TiO}_2$ .<sup>20–25</sup> Liu *et al.*, used a  $\text{WO}_3/\text{Sb}_2\text{S}_3$  heterojunction electrode and obtained four-fold enhancement of photocurrent density ( $1.79 \text{ mA cm}^{-2}$  at 0.8 V) over pristine  $\text{WO}_3$ .<sup>26</sup> Also, Song *et al.* reported a current density of  $0.79 \text{ mA cm}^{-2}$  at an onset of  $-0.08 \text{ V}$  for  $\text{TiO}_2/\text{Sb}_2\text{S}_3$ , while pristine  $\text{TiO}_2$  shows poor performance under similar experimental conditions.<sup>27</sup> Nonetheless, such binary electrodes pose issues such as slow injection of electrons and high rates of carrier recombination. The efficiency of electron injection is improved by interfacing with noble metal nanoparticles (NPs) such as Ag and Au.<sup>28–34</sup> Here the metal NP not only lowers the rate of energy-wasting charge recombination in semiconductors but also enhances the photoreactivity due to the localized surface plasmon resonance (LSPR).<sup>35–38</sup> The collectively oscillating conduction band electrons in a plasmonic NP facilitate the absorption of a wide range of solar photons, which depends upon the metal

<sup>a</sup>Department of Chemistry, Indian Institute of Technology, Kandi, Hyderabad, India-502285<sup>b</sup>Center for Emergent Functional Matter Science, National Chiao Tung University, Taiwan<sup>c</sup>Research Institute for Electronic Science, Hokkaido University, N20 W10, Sapporo, Hokkaido 001-0020, Japan. E-mail: [biju@es.hokudai.ac.jp](mailto:biju@es.hokudai.ac.jp)

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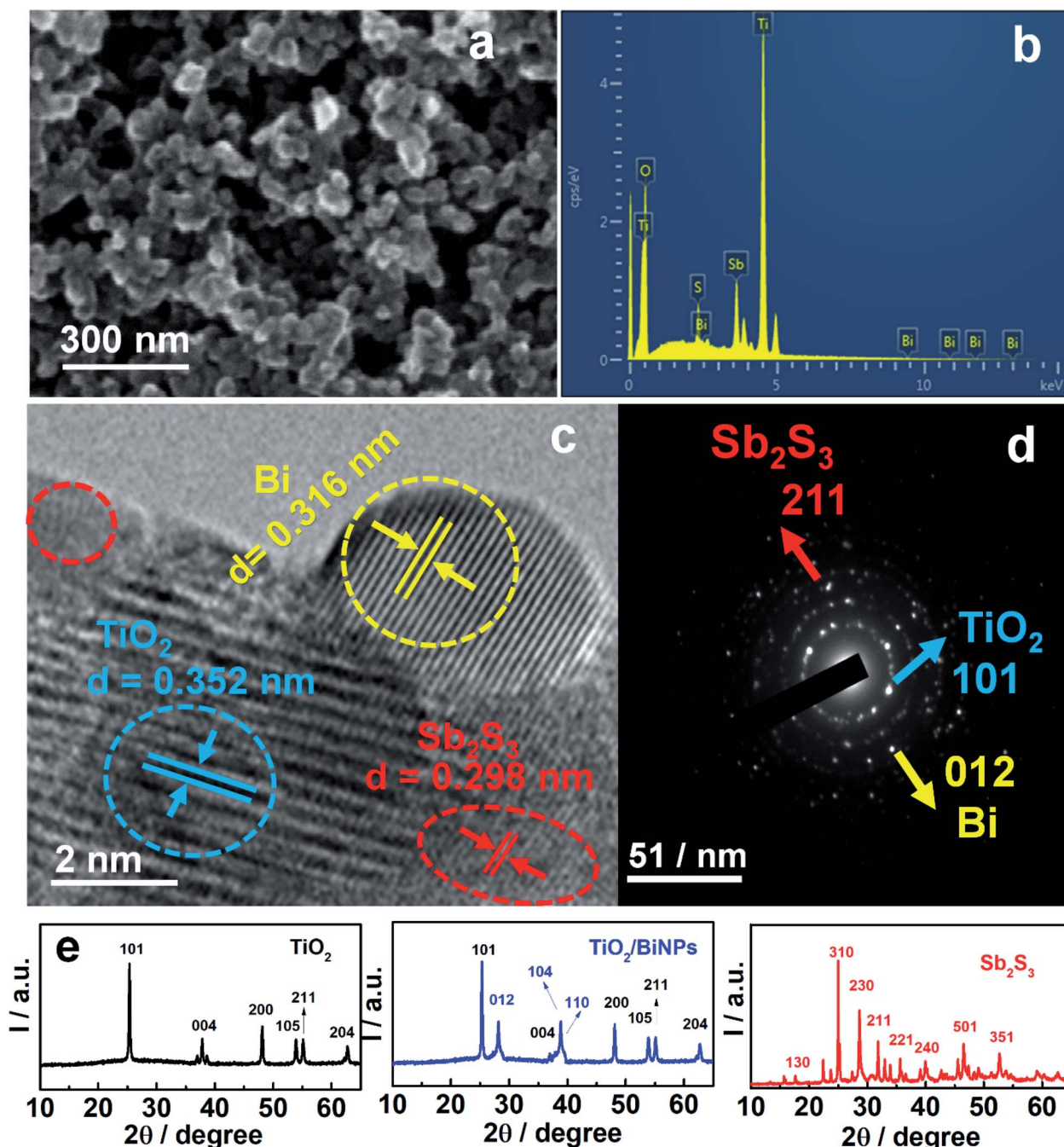


Fig. 1 Characterization of the ternary photoanode and anode materials. (a) FESEM image, (b) EDAX image, (c) HR-TEM image and (d) SAED pattern of  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$ , and (e) XRD patterns of  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{BiNP}$  and  $\text{Sb}_2\text{S}_3$ .

as well as the size and shape of the NP. As an example,  $\text{CdS}/\text{Ag}/\text{ZnO}$  photoanodes show higher (3.13%) solar energy conversion efficiency than  $\text{CdS}/\text{ZnO}$  alone (1.35%).<sup>39</sup> Also, Li *et al.*, fabricated  $\text{CdS}/\text{Au}/\text{TiO}_2$  photoanodes and achieved photocurrent density as high as  $4.07 \text{ mA cm}^{-2}$  at 0 V vs. the  $\text{Ag}/\text{AgCl}$  electrode.<sup>40</sup> Although bismuth (Bi) NPs expand the optical absorption window and suppress the rate of carrier recombination in conventional semiconductors,<sup>41–46</sup> the plasmonic properties of this earth-abundant NP<sup>41</sup> are yet to be explored in PEC applications.

We report the ternary photoanode  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  and its application in high efficiency PEC water splitting. The electrode is prepared by chemical bath deposition (CBD). We accomplish an order of magnitude higher efficiency of PEC hydrogen evolution by supplementing the conventional  $\text{TiO}_2$  electrode with the narrow bandgap  $\text{Sb}_2\text{S}_3$  and the plasmonic properties of BiNPs. The ternary electrode provides us with a photocurrent density as high as  $4.21 \text{ mA cm}^{-2}$  at 1.23 V, the incident photon to current conversion efficiency (IPCE) at 31%, and solar power conversion efficiency at 3.85%. We hypothesize that BiNPs



enhance the open circuit voltage by bending the band across the junctions, expediting the transfer of photogenerated electrons from  $\text{Sb}_2\text{S}_3$  to  $\text{TiO}_2$ , and increase the short circuit current by the SPR effect.

The ternary photoanode  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  was prepared by the doctor-blade technique and chemical bath deposition. The details of photoanode fabrication are provided in the Experimental section. The ternary electrode and BiNPs were characterized by FESEM, EDAX, HRTEM, Raman spectroscopy and XRD studies. Fig. 1a shows the surface morphology of the photoanode  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$ , which aggregate randomly with no specific shape displaying high roughness. Further, the EDAX image confirms that the constituent elements in the ternary electrode are Ti, O, Bi, Sb and S (Fig. 1b) and it reveals that BiNPs are effectively incorporated on  $\text{TiO}_2$  in the composite. The SEM and EDAX images of BiNPs,  $\text{TiO}_2/\text{BiNP}$  and  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  are shown in Fig. S1.† Further, the HRTEM image of the  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoanode reveals the placement of BiNPs in between  $\text{TiO}_2$  and  $\text{Sb}_2\text{S}_3$  and the lattice fringes of individual components (Fig. 1c). Body centered tetragonal  $\text{TiO}_2$  has a lattice spacing of 0.352 nm that corresponds to (101) (JCPDS-894921), while orthorhombic  $\text{Sb}_2\text{S}_3$  has a lattice constant of 0.298 nm that is assigned to the (211) plane (JCPDS-421393). Plasmonic BiNPs have a lattice spacing of 0.316 nm which is assigned to the (012) plane (JCPDS-851331). The selected area electron diffraction (SAED) patterns (Fig. 1d) confirm the crystalline structures of BiNP,  $\text{TiO}_2$  and  $\text{Sb}_2\text{S}_3$ . The TEM images of pure BiNPs show a spherical shape with an average particle size of around 5–10 nm (Fig. S2a†) and the lattice spacing of 0.32 nm relates to the (012) plane of Bi (JCPDS-851331) (Fig. S2b†). The XRD pattern of pure BiNPs is shown in the ESI (Fig. S2c†) and BiNPs possess a rhombohedral structure with the (012), (104), (110) and (202) planes corresponding to  $d$ -spacings of 3.25, 2.34, 2.24 and 1.83 Å respectively (JCPDS: 851331). No peaks of other bismuth compounds (such as  $\text{Bi}_2\text{O}_3$ ) can be observed in the pattern which suggests that  $\text{Bi}^{3+}$  ions reduced to metallic Bi.<sup>42,44,49</sup> The XRD patterns of  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{BiNP}$  and  $\text{Sb}_2\text{S}_3$  are shown in Fig. 1e, which demonstrate the body-centered tetragonal crystal lattice of  $\text{TiO}_2$ , rhombohedral structure of BiNPs and orthorhombic primitive lattice of  $\text{Sb}_2\text{S}_3$ .  $\text{TiO}_2$  has a body-centered tetragonal crystal lattice with the (101), (103), (004), (200), (105), (211) and (204) planes corresponding to  $d$ -spacings of 3.51, 2.43, 2.36, 1.89, 1.69, 1.66, 1.47 and 1.36 Å respectively (JCPDS-894921). In  $\text{TiO}_2/\text{BiNP}$ , the presence of (012), (104) and (110) planes, confirms formation of  $\text{TiO}_2/\text{BiNPs}$ .  $\text{Sb}_2\text{S}_3$  has an orthorhombic primitive lattice with the (130), (310), (230), (211), (221), (240), (501) and (351) planes corresponding to  $d$ -spacings of 3.54, 3.49, 3.16, 3.01, 2.74, 2.55, 1.93 and 1.73 Å respectively (JCPDS: 421393). Raman spectral analysis of BiNPs is shown in Fig. S2d,† where pure BiNPs show two characteristic vibrational bands at 68 and 91  $\text{cm}^{-1}$  corresponding to  $E_g$  and  $A_{1g}$  modes respectively. Furthermore, the Raman spectra of  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{BiNP}$  and  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  (Fig. S3†) help us to clarify the structures of the electrode components. The anatase phase of  $\text{TiO}_2$  was confirmed by the Raman bands at 144 and 635  $\text{cm}^{-1}$  that correspond to the  $E_g$  mode while 396 and 512  $\text{cm}^{-1}$  belong to  $B_{1g}$  and  $B_{1g}/A_{1g}$  modes respectively. In  $\text{TiO}_2/\text{BiNP}$ , the

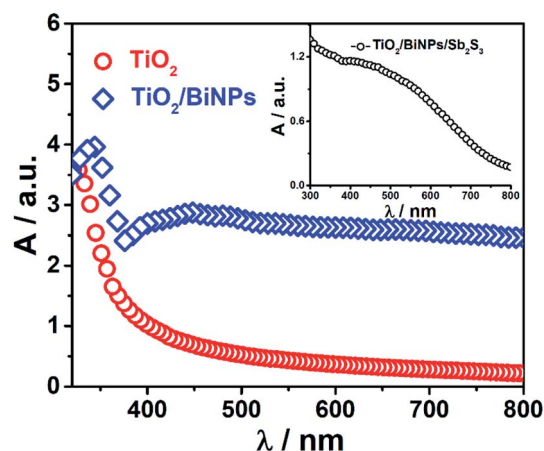


Fig. 2 Absorption spectra of the  $\text{TiO}_2$  and  $\text{TiO}_2/\text{BiNP}$  electrodes. Inset: absorption spectrum of the  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  electrode.

presence of peaks at 68 and 91  $\text{cm}^{-1}$ , confirms that BiNPs are well incorporated with  $\text{TiO}_2$  particles. Further, the introduction of  $\text{Sb}_2\text{S}_3$  onto  $\text{TiO}_2/\text{BiNP}$ , shows peaks at 135, 188 and 431  $\text{cm}^{-1}$  that belong to the vibrational modes of  $\text{SbS}_3$  units and Sb–Sb bonds in  $\text{Sb}_2\text{S}_3$  respectively. The Raman spectra of a pure  $\text{Sb}_2\text{S}_3$  sample are displayed in Fig. S4.†

With the modification of the  $\text{TiO}_2$  electrode using BiNPs and  $\text{Sb}_2\text{S}_3$  sensitizers, the absorption band of the electrode extends from the UV to NIR region. Fig. 2 shows the absorption spectra of  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{BiNP}$  and  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  electrodes, which reveal the strong absorption of light in the range of 400–800 nm by BiNPs. This corresponds to near UV to NIR absorption, which is confirmed by the absorption spectra of an aqueous solution of pure BiNPs (Fig. S5a and b†). The broad absorption in the UV-Vis-NIR region (Fig. 2) is due to the SPR of BiNPs.<sup>49,50</sup> The  $\text{TiO}_2$  electrode selectively absorbs light with a wavelength smaller than 390 nm. The absorption onset of  $\text{TiO}_2$  is expanded to the 400–750 nm region with the chemical bath deposition of the small bandgap  $\text{Sb}_2\text{S}_3$  (Fig. S4b†). The bandgaps of  $\text{TiO}_2$  and  $\text{Sb}_2\text{S}_3$  employed in this work are estimated at 3.13 and 1.6 eV, respectively.

To evaluate the plasmonic effect of BiNPs on the PEC activity of the ternary electrode, we examined the photocurrent response by each photoanode ( $\text{TiO}_2$  alone,  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  or  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$ ) by linearly sweeping the voltage (LSV plots) in a three-electrode system (Ag/AgCl as the reference electrode and Pt as the counter electrode), and the results are summarized in Fig. 3a. The pH of the electrolyte was set at 12.7 and the cells are illuminated with simulated sunlight (AM1.5). All the potentials reported against the reversible hydrogen electrode (RHE) are calculated using the Nernst equation. Upon chopped light illumination, the photocurrent density values for  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{BiNP}$ ,  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  and  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoanodes are 0.15, 0.61, 1.83 and 4.21  $\text{mA cm}^{-2}$ , respectively, at 1.23 V. The highest photo-response in terms of current density is observed for  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  at the lowest onset potential of 0.13 V. The enhanced photocurrent density of the  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  electrode, which is promising for PEC water splitting, is attributed





to the ability of  $\text{Sb}_2\text{S}_3$  to absorb the visible-NIR light and the SPR of BiNPs. In addition, the efficiency of a photoelectrode is expressed in terms of solar to hydrogen conversion (STH). The STH conversion efficiencies for  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{BiNP}$ ,  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  and  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  are estimated at 0.03, 0.22, 0.51 and 1.56% at 0.6 V (Fig. 3b), which are calculated from short circuit current density and the intensity of incident light ( $\text{ESI}^\dagger$ ). The  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoanode provides us with 3- and 15-times higher STH conversion efficiency than  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  and  $\text{TiO}_2$ , respectively. Furthermore, estimation of the PEC water splitting performance of the  $\text{TiO}_2/\text{Sb}_2\text{S}_3/\text{BiNP}$  electrode was carried out by using LSV plots and the results are presented in Fig. S6a.<sup>†</sup> Upon chopped light illumination, the  $\text{TiO}_2/\text{Sb}_2\text{S}_3/\text{BiNP}$  electrode showed a maximum photocurrent density of  $3.16 \text{ mA cm}^{-2}$  at 1.23 V. Further, the high STH conversion efficiency of the  $\text{TiO}_2/\text{Sb}_2\text{S}_3/\text{BiNP}$  electrode is 0.68% at 0.6 V (Fig. S6b<sup>†</sup>). Therefore, the  $\text{TiO}_2/\text{Sb}_2\text{S}_3/\text{BiNP}$  electrode provides lower photocurrent density and STH efficiency than the  $\text{TiO}_2/\text{Bi}/\text{Sb}_2\text{S}_3$  electrode. These results suggested that BiNP-embedded  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  can be better understood based on the staggered energy

alignment of the  $\text{TiO}_2/\text{Bi}/\text{Sb}_2\text{S}_3$  compared to BiNPs grown on the surface of  $\text{Sb}_2\text{S}_3$  ( $\text{TiO}_2/\text{Sb}_2\text{S}_3/\text{BiNP}$ ) for improving PEC water splitting.

The stability of the photocurrent response by  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{BiNP}$ ,  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  and  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoanodes is examined by chronoamperometric ( $I-t$  curve) studies. Also, the photocurrent responses of the electrodes are tested at 1.23 V *versus* RHE (0.6 *versus* Ag/AgCl) under simulated solar (AM1.5) irradiation, and the results are presented in Fig. 3c. The long-term stability of the ternary electrode is noteworthy; only little decay in performance is observed after 5100 s.  $\text{H}_2$  evolution is quantified using gas chromatography and the amount of  $\text{H}_2$  evolved is plotted as a function of time at 1.23 V *vs.* RHE. As shown in Fig. 3d, the  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoanode provides maximum (3.2 mmol)  $\text{H}_2$  evolution over 2 h, which is when compared to  $\text{TiO}_2/\text{Sb}_2\text{S}_3$ . The high photocurrent efficiency and  $\text{H}_2$  evolution performances of  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  are in good agreement.

The ability of photoelectrodes to convert the incident photons into electrons is examined by IPCE experiments in

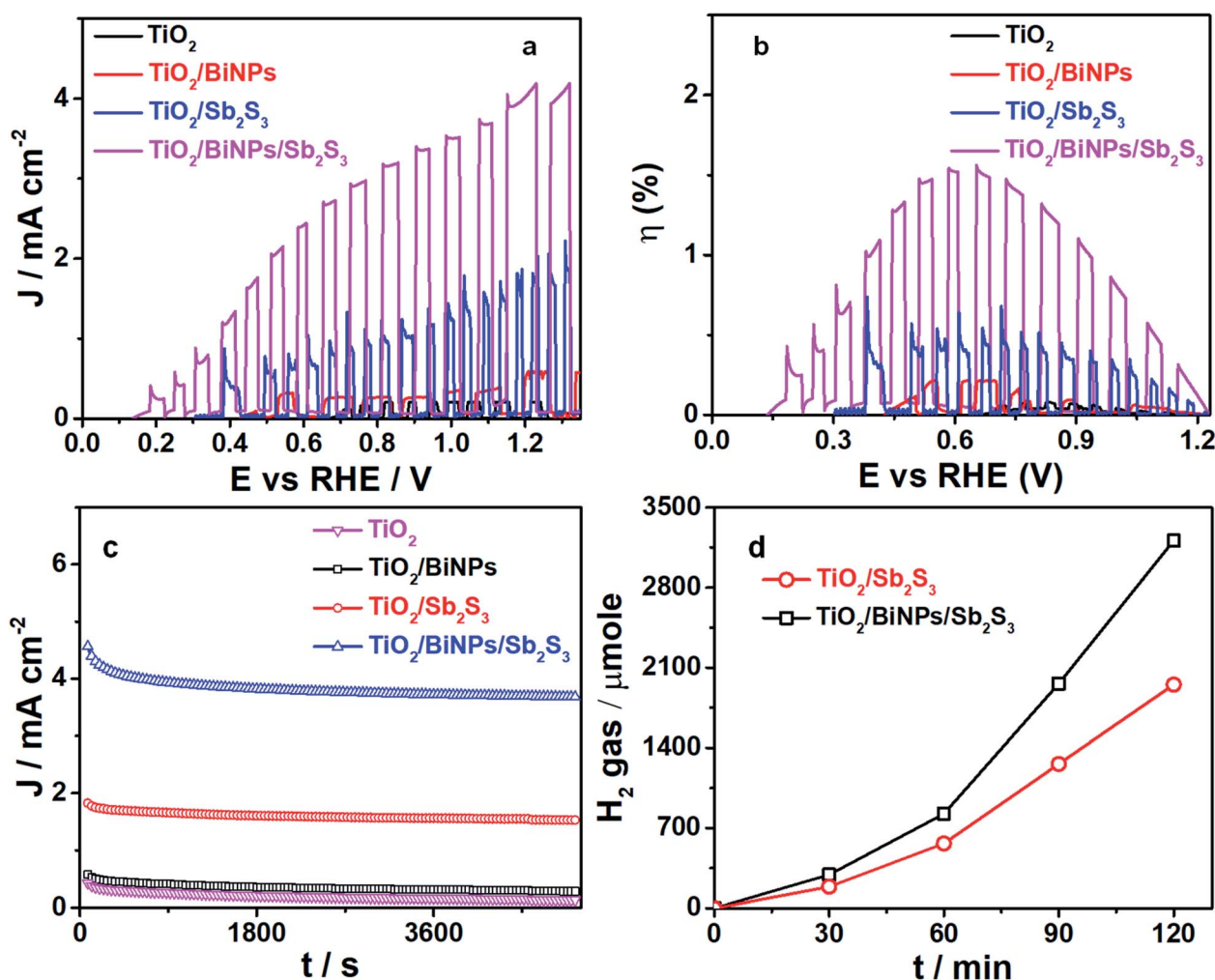


Fig. 3 PEC properties of photoanodes. (a) LSV plots, (b) solar to hydrogen conversion efficiency and (c) photocurrent stability plots of  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{BiNP}$ ,  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  and  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  electrodes. Here 0.1 M  $\text{Na}_2\text{SO}_3$  and 0.1 M  $\text{Na}_2\text{SO}_4$  are used as the electrolyte solution under simulated solar irradiation. (d) Temporal evolution of  $\text{H}_2$  gas by  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  and  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  electrodes.



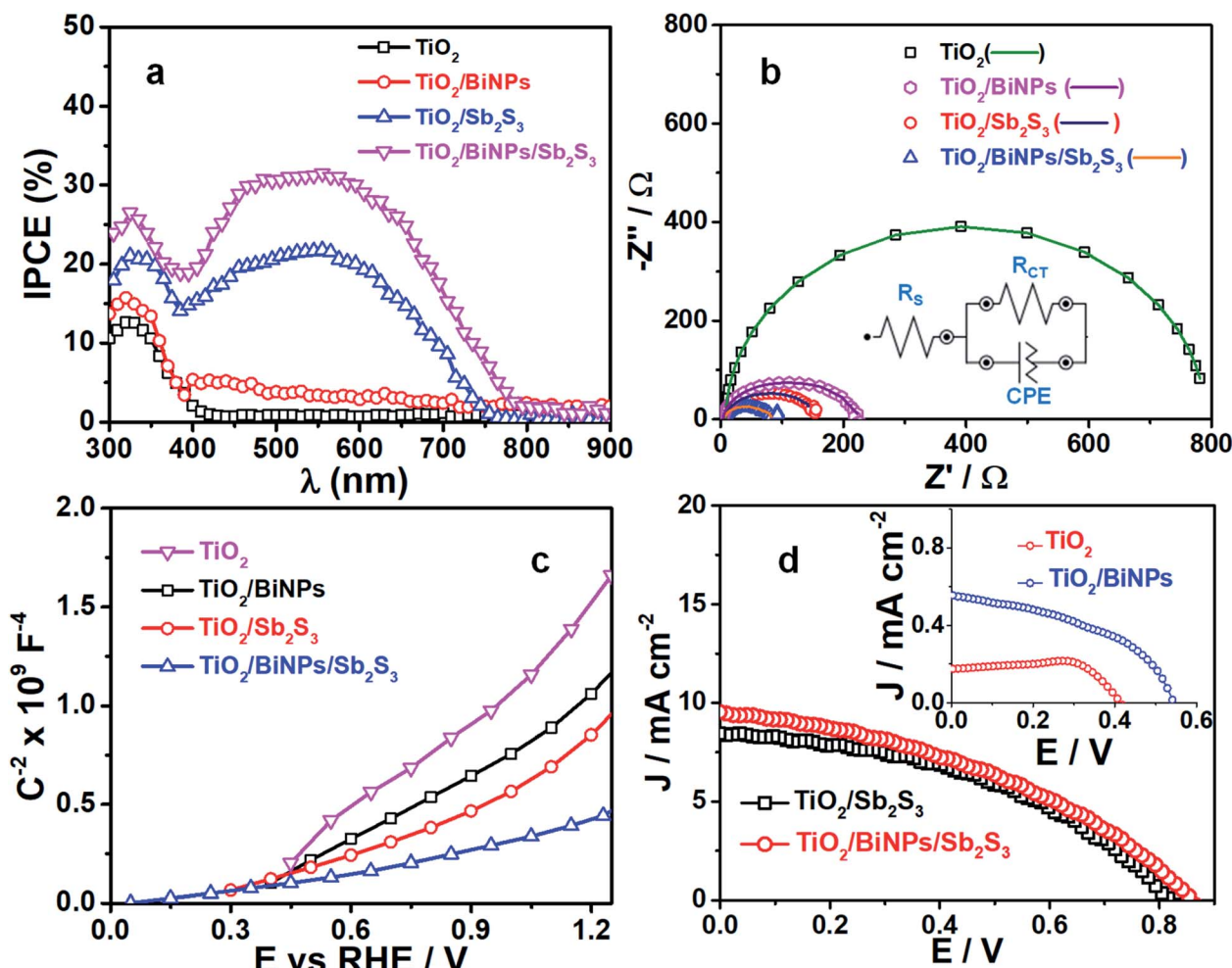


Fig. 4 IPCE and  $I/V/R$  characteristics of photoanodes. (a) Wavelength-dependent IPCE plots (b) Nyquist plots (symbol) with an equivalent circuit (dot lines) and (c) Mott–Schottky plots of pristine  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{BiNP}$ ,  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  and  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoanodes. (d)  $J-V$  data of  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  and  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoanodes.

a two-electrode system, but without applying any external bias, as shown in Fig. 4a. The maximum IPCE for  $\text{TiO}_2$  is 13% in the UV region (300–400 nm), which is negligibly small in the visible to NIR region (400–900 nm). We find an increase in the IPCE for the  $\text{TiO}_2$  electrode supplemented with BiNPs, which is 15% in the UV region (300–400 nm) and additional contribution of 2–5% in the 400–800 nm region. This enhancement in IPCE performance by BiNPs is attributed to their SPR in the visible to NIR region. When compared to the 13% IPCE of  $\text{TiO}_2$  alone, the  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  electrode shows *ca.* 20% IPCE in the 300–750 nm region, which is attributed to the small bandgap of  $\text{Sb}_2\text{S}_3$ . Thus, by combining the plasmonic effect of BiNPs with the small bandgap of  $\text{Sb}_2\text{S}_3$  in  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$ , we accomplish 31% IPCE for the ternary photoanode. Also, this result implies that BiNPs in the heterojunction suppress undesired recombination of photogenerated charge carriers in  $\text{Sb}_2\text{S}_3$ .

The kinetics of charge injection at the photoanode/electrolyte interfaces are studied by measuring the resistance across the electrode/electrolyte interface. Here impedance measurements were carried out under the light illumination of

$\text{TiO}_2$ ,  $\text{TiO}_2/\text{BiNP}$ ,  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  or  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  electrodes immersed in a solution of 0.1 M  $\text{Na}_2\text{SO}_3$  and 0.1 M  $\text{Na}_2\text{SO}_4$  (1 : 1, v/v) and the plots were fitted into the equivalent circuits shown in the inset in Fig. 4b, where  $R_s$ ,  $R_{CT}$  and CPE represent the electrolyte solution resistance, interfacial charge transfer resistance at electrode/electrolyte and constant phase element for photoelectrode/electrolyte, respectively. The fitted parameters are given in Table 2 (ESI†). All the electrodes have solution resistance ( $R_s$ ) in the range of 4–9  $\Omega$ . The high  $R_{CT}$  value of 775  $\Omega$  for  $\text{TiO}_2$  is due to low charge separation and the lowest  $R_{CT}$  value of 71  $\Omega$  is observed for the  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoanode compared to other electrodes, indicating the facile transfer of charge carriers at the  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoanode/electrolyte interface, which is consistent with the excellent PEC performance of the electrode. Also, these charge injection properties are consistent with the electron lifetime ( $\tau$ ) values estimated from Bode plots (Fig. S7†). The  $\tau$  values are estimated using the relation  $\tau = 1/(2\pi f_{\max})$  where  $f_{\max}$  is the frequency maxima. The  $\tau$  values calculated for pristine  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{BiNP}$ ,  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  and  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  films are 1.13, 1.22, 1.46 and 1.65 ms,



respectively. The large  $\tau$  value corresponds to a high rate of carrier transport or low rate of undesired carrier recombination in the  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoelectrode, which is attributed to the effect of BiNPs. In other studies, BiNPs increase the  $\tau$  value of  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$ , which is in good agreement with the higher photocurrent density. Furthermore, we construct the Mott-Schottky plots of electrodes (Fig. 4c), where the linear profiles with positive slopes confirm the n-type characteristics of the photoanodes.

To evaluate the role of plasmonic BiNPs on the power conversion efficiency of the ternary electrode, we constructed different photoelectrochemical solar cells and recorded the current-voltage ( $J$ - $V$ ) characteristics of the cells under 1 sun illumination ( $100 \text{ mW cm}^{-2}$ ). Here the electrolyte used was polysulfide ( $0.1 \text{ M Na}_2\text{S}$  and  $0.1 \text{ M S}$ ) solution. The active area of the working electrode ( $\text{TiO}_2$ ,  $\text{TiO}_2/\text{Sb}_2\text{S}_3$ , or  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$ ) was set at  $0.15 \text{ cm}^2$ , and the counter electrode used was C-fabric. The  $J$ - $V$  responses of the cells are shown in Fig. 4d and the solar cell parameters are summarized in Table S1 (ESI†). The  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  cell shows an open circuit voltage ( $V_{\text{OC}}$ ) of  $\sim 860 \text{ mV}$  and a short circuit current density ( $J_{\text{SC}}$ ) of  $9.54 \text{ mA cm}^{-2}$ , which are greater than those of the  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  electrode. The power conversion efficiency (PCE) of cells with  $\text{TiO}_2$ ,  $\text{TiO}_2/\text{Bi}$ ,  $\text{TiO}_2/\text{Sb}_2\text{S}_3$  and  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  electrodes is 0.019, 0.39, 2.99 and 3.85% respectively. The  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  based cells achieved the highest PCE and  $V_{\text{OC}}$  was also the highest for this cell compared to the other electrodes. The high value of  $V_{\text{OC}}$  is attributed to the bending of bands and improvement of charge separation induced by BiNPs. We attribute the high  $J_{\text{SC}}$  value to the SPR effect of BiNPs, which increases the carrier density. These NPs not only transfer the photogenerated electrons from  $\text{Sb}_2\text{S}_3$  to FTO *via*  $\text{TiO}_2$  but also act as photosensitizers, which are obvious from the high  $J_{\text{SC}}$  and  $V_{\text{OC}}$  values of FTO/ $\text{TiO}_2$ /BiNP when compared with FTO/ $\text{TiO}_2$ . The  $J$ - $V$  characteristics FTO-based electrodes are shown in the inset in Fig. 4d. Furthermore, the higher PCE (0.39%) of the FTO/ $\text{TiO}_2$ /BiNP electrode, which is when compared with FTO/ $\text{TiO}_2$  (0.01%), suggests that BiNPs transfer photo-generated electrons to the external circuit. In addition, the  $\text{TiO}_2/\text{Bi}$  electrode shows high  $J_{\text{SC}}$  ( $0.56 \text{ mA cm}^{-2}$ ) and  $V_{\text{OC}}$  ( $0.54$

V) compared to the pure  $\text{TiO}_2$  film which is due to different regions of the solar spectrum being absorbed and converted to current.

The Fermi level of BiNPs, and conduction (CB) and valence band (VB) positions of  $\text{TiO}_2$  and  $\text{Sb}_2\text{S}_3$  are obtained from cyclic voltammetry (Fig. S8†). The procedure for calculations of band positions in the energy level diagram is as shown in the ESI.† The mechanism of PEC performance of the ternary photoanode  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  is schematically presented in Fig. 5. Here the simulated solar illumination of the electrode leads to the generation of electron/hole pairs in the conduction (CB)/valence bands (VB) of  $\text{Sb}_2\text{S}_3$ . The photogenerated holes in the VB of  $\text{Sb}_2\text{S}_3$  react with the sacrificial reagent ( $\text{SO}_3^{2-}$ ) and hydroxyl ion to produce  $\text{SO}_4^{2-}$  and oxygen. On the other hand, BiNPs assist the transport of photogenerated electrons through the external circuit to the counter electrode (Pt) where protons are reduced to liberate hydrogen. In this hypothetical mechanism, the primary role of BiNPs is to expedite the transfer of electrons from the conduction band of  $\text{Sb}_2\text{S}_3$  to that of  $\text{TiO}_2$  and finally to the back contact FTO.

In summary, by combining the low bandgap  $\text{Sb}_2\text{S}_3$  with the plasmonic BiNPs in the ternary  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoanode, we find that the efficiencies of photocurrent generation and photoelectrochemical hydrogen evolution increase by an order of magnitude when compared with the pristine  $\text{TiO}_2$  electrode. This ternary photoanode shows photocurrent density as high as  $4.21 \text{ mA cm}^{-2}$  at  $1.23 \text{ V}$  with a solar to hydrogen conversion efficiency of 1.56% at  $0.6 \text{ V}$ . BiNPs deliver the SPR effect to  $\text{Sb}_2\text{S}_3$  and increase the rate of electron transport from  $\text{Sb}_2\text{S}_3$  to  $\text{TiO}_2$ , and as a result, the maximum of IPCE becomes 31% and the solar power conversion efficiency becomes 3.85%. By expediting the transfer of photogenerated electrons from  $\text{Sb}_2\text{S}_3$  to the  $\text{TiO}_2$  electrode, BiNPs increase the open circuit voltage up to  $860 \text{ mV}$ . Also, with the plasmonic nearfield effect, the short circuit current reaches up to  $9.54 \text{ mA cm}^{-2}$ . Overall, by combining the earth-abundant and cost-effective plasmonic BiNPs and the narrow bandgap  $\text{Sb}_2\text{S}_3$  on the  $\text{TiO}_2$  electrode, we develop a stable, ternary photoanode and accomplish high efficiency photocurrent generation and hydrogen evolution.

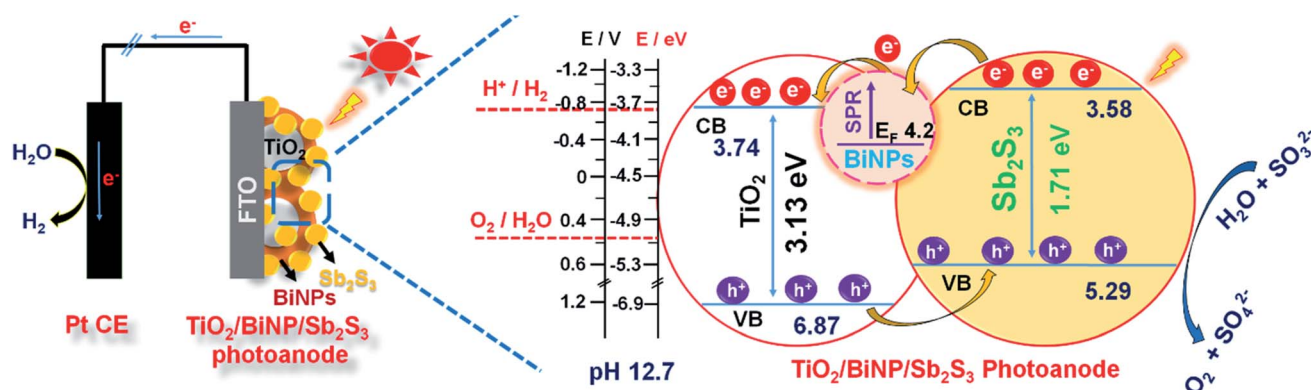


Fig. 5 A scheme of a PEC cell constructed using the ternary photoanode  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$ , showing the band alignment, charge transport and redox reactions under the simulated solar irradiation.





## Experimental

### Materials and samples

Antimony chloride ( $\text{SbCl}_3$ ), sodium thiosulfate ( $\text{Na}_2\text{S}_2\text{O}_3$ ), bismuth nitrate pentahydrate [ $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$ ], dimethyl sulfoxide (DMSO), sodium sulfate ( $\text{Na}_2\text{SO}_4$ ), elemental sulfur, methanol and acetone were obtained from Aldrich. Titanium tetrachloride ( $\text{TiCl}_4$ ), sodium borohydride ( $\text{NaBH}_4$ ), sodium sulfide ( $\text{Na}_2\text{S}$ ), sodium sulfite ( $\text{Na}_2\text{SO}_3$ ) and tannic acid were obtained from Merck.  $\text{TiO}_2$  paste (18NR-T) used for electrode preparation was obtained from Dyesol. Deionized water (resistivity  $\sim 18.2 \Omega \text{ cm}$ ) was obtained from a Millipore Direct-Q3 UV system. Fluorine doped tin oxide (FTO) glass (sheet resistance  $\sim 25 \Omega \text{ cm}^{-2}$ ) was obtained from Pilkington.

### Preparation of photoanodes

The  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  ternary photoanode was prepared by the doctor-blade technique. The photoanode was fabricated as follows:  $\text{TiO}_2$  paste was coated onto the pre-cleaned FTO substrates by the doctor-blade method, which was followed by annealing the  $\text{TiO}_2$  film at  $500^\circ\text{C}$  for 30 min to provide a thin layer of  $\text{TiO}_2$  on FTO. Further, the  $\text{TiO}_2$  film was immersed in a hot ( $70^\circ\text{C}$ ) solution (40 mM in isopropanol) of  $\text{TiCl}_4$  for 30 min. The film was rinsed with DI water and annealed at  $500^\circ\text{C}$  for 30 min to provide the uniform  $\text{TiO}_2$  electrode. Further, BiNPs were grown onto the  $\text{TiO}_2$  electrode. In brief,  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  was dissolved in DMSO and a solution containing a mixture of  $\text{NaBH}_4$  and tannic acid was slowly added, which brought about the reduction of  $\text{Bi}^{3+}$  to  $\text{Bi}^0$  and the obtained BiNPs were dispersed in water for further studies. Finally, the  $\text{TiO}_2$  film was vertically introduced into the bismuth precursor solution and sonicated for 15 min to form the  $\text{TiO}_2/\text{BiNP}$  film, which was brown in color. Further,  $\text{Sb}_2\text{S}_3$  was grown onto the  $\text{TiO}_2/\text{BiNP}$  film by the chemical bath deposition process to form the ternary electrode  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$ .<sup>47,48</sup> Briefly, 400 mg of  $\text{SbCl}_3$  was dissolved in 2.5 mL of acetone, which was followed by the addition of 72.5 mL of deionized water to yield a turbid solution that subsequently converted into a white precipitate. Subsequently, 3.95 g of  $\text{Na}_2\text{S}_2\text{O}_3$  dissolved in 25 mL of water was added to the  $\text{SbCl}_3$  solution under stirring. The  $\text{TiO}_2/\text{BiNP}$  film was vertically dipped in the  $\text{Sb}_2\text{S}_3$  solution at  $25^\circ\text{C}$  for 60 min, which was followed by rinsing the electrode with deionized water and drying on a hot plate at  $60^\circ\text{C}$ . Finally, the film was annealed at  $300^\circ\text{C}$  for 15 min under an argon atmosphere to yield the dark-orange colored  $\text{TiO}_2/\text{BiNP}/\text{Sb}_2\text{S}_3$  photoanode. Preparation of the electrode is illustrated in the ESI.†

### Characterization techniques

The surface morphologies of the samples were analyzed using a field emission scanning electron microscope (FESEM-Zeiss supra 40). High-resolution transmission electron microscope (HR-TEM) images were recorded using a TECNAI G-2 FEI instrument operating at 300 kV. The powder X-ray diffraction (XRD) patterns of samples were measured on a PANalytical, Xpert PRO instrument. The Raman spectra of photoanode

materials were recorded by using a Bruker Senterra dispersive Raman microscope spectrometer with an excitation wavelength of 532 nm. The absorption spectra of the synthesized electrodes were recorded on a UV-Vis spectrophotometer (Shimadzu UV-3600). PEC measurements were performed in an aqueous electrolyte containing a 1 : 1 mixture of 0.1 M  $\text{Na}_2\text{SO}_3$  and 0.1 M  $\text{Na}_2\text{SO}_4$ , using a three electrode system with Pt and Ag/AgCl as the counter and reference electrodes, respectively. The current *versus* potential ( $I$ - $V$ ) characteristics of photoanodes were recorded using a LOT-Oriel-Autolab with a 150 W xenon arc lamp with  $100 \text{ mW cm}^{-2}$  intensity. Chronoamperometric ( $I$ - $t$ ) plots, Mott-Schottky plots, cyclic voltammetry (CV) and electrochemical impedance spectra (EIS) were measured on an Autolab PGSTAT 302N equipped with NOVA 1.9 software. The irradiance intensity was confirmed by calibrating with a  $2 \text{ cm} \times 2 \text{ cm}$  Si Reference Cell and re-affirmed with a radiant power meter from Newport (842-PE). The IPCE was measured using a quantum efficiency measurement system (Oriel IQE-200), where 250 W quartz tungsten and halogen lamps were used as the light sources, the monochromator path length was 1/8M and the spot size was  $1 \text{ mm} \times 2.5 \text{ mm}$  rectangular at focus. This instrument gave the IPCE values as a function of wavelength directly.

## Conflicts of interest

There are no conflicts to declare.

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