

Cite this: *Nanoscale Adv.*, 2020, 2, 4539

# Tuning the binding behaviors of a protein YAP65WW domain on graphenic nano-sheets with boron or nitrogen atom doping†

Xiao Jia,<sup>a</sup> Yanmei Yang,<sup>id</sup><sup>\*b</sup> Yang Liu,<sup>a</sup> Weihua Niu,<sup>b</sup> Yong-Qiang Li,<sup>id</sup><sup>a</sup> Mingwen Zhao,<sup>id</sup><sup>a</sup> Yuguang Mu,<sup>id</sup><sup>\*c</sup> and Weifeng Li,<sup>id</sup><sup>\*a</sup>

In recent years, nanomaterials have attracted considerable research attention for biological and medical related applications due to their well-recognized physical and chemical properties. However, the deep understanding of the binding process at the protein–nanomaterial interface is essential to solve the concern of nano-toxicity. Here, we study the interactions between the recently reported graphenic nano-sheets, BC<sub>3</sub> and C<sub>3</sub>N, and a prototypical protein (YAP65WW domain) *via* atomistic molecular dynamics simulations. Our simulations reveal that elemental doping is an effective way to tune the binding characteristics of YAP65WW with two nanomaterials. While YAP65WW can be attracted by two nanomaterials, the BC<sub>3</sub> sheet is less able to disrupt the protein structure than C<sub>3</sub>N. From the energy point of view, this is because protein residues demonstrate a binding preference with the trend from electron rich nitrogen to electron deficient boron. Structural analyses of the bio-nano interface revealed the formation of an ordered water shell on the BC<sub>3</sub> surface, which was compatible to the crystal pattern of BC<sub>3</sub>. When a protein binds with BC<sub>3</sub>, these interfacial water molecules protect the protein from being disrupted. We suggest that elemental doping is efficient to produce fruitful biological-effects of graphenic nanomaterials, which make it a prospective solution for the future design and fabrication of advanced nanomaterials with desired function.

Received 5th May 2020  
Accepted 25th August 2020

DOI: 10.1039/d0na00365d

rsc.li/nanoscale-advances

## 1. Introduction

In recent years, graphene has attracted considerable research interest in numerous fields such as electronic<sup>1,2</sup> and optical materials,<sup>3,4</sup> catalytic agents, sensors and biomedical materials<sup>5</sup> due to its well-recognized physical and chemical properties. Particularly, in biological and medical applications, graphene has been widely studied in tumor therapy,<sup>6</sup> biosensors<sup>7</sup> and gene sequencing platforms.<sup>8,9</sup> However, the major concern of the potential nano-toxicity has been attracting enormous attention of researchers worldwide. Numerous studies have shown that graphene can cause severe damage to bio-macromolecules, such as protein unfolding, DNA unwinding, and destructive extraction of the cell membrane.<sup>10–14</sup> For instance, our previous theoretical simulations demonstrate that

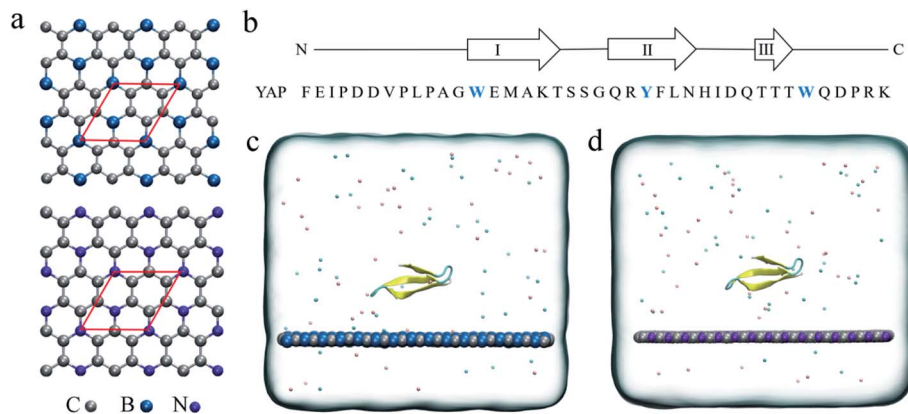
when a graphene monolayer has defects, the protein unfolding speed is significantly accelerated by specific interactions between defects and charged residues.<sup>15,16</sup> To reduce the toxicity, surface morphology engineering<sup>17–19</sup> and functionalization with polyethylene glycol or serum proteins have been demonstrated to be a success.<sup>20</sup> Meanwhile, our previous studies have demonstrated that a nitrogenized graphene (abbreviated as C<sub>2</sub>N) monolayer is biocompatible to both protein and DNA systems.<sup>21,22</sup> Thus, modifying the graphenic structures by elemental doping is expected to be an efficient way to tune the biological functions of the nanomaterials.

Recently, the successful experimental syntheses of boron-doped graphene, BC<sub>3</sub> (ref. 23) and nitrogen-doped graphene, C<sub>3</sub>N<sup>24</sup> have been achieved. BC<sub>3</sub> and C<sub>3</sub>N monolayers have excellent material chemistry properties and are used in batteries<sup>25–27</sup> and high sensitive gas sensors.<sup>28–32</sup> Structurally, the B and N atoms are arranged in a patterned manner in graphenic nanosheets. Each B or N atoms is covalently connected to three C atoms (Fig. 1a). Due to the electronegativity difference between the three constitute elements, B, C and N, there is an intrinsic electron transfer within the planar structure of BC<sub>3</sub> and C<sub>3</sub>N, resulting in periodically distributed dipoles. A recent theoretical study on the interactions of BC<sub>3</sub> and water clearly showed that BC<sub>3</sub> induced a significant re-arrangement of water dipoles at the interface, making BC<sub>3</sub> a promising candidate for

<sup>a</sup>School of Physics, State Key Laboratory of Crystal Materials, Shandong University, Jinan, Shandong, 250100, China. E-mail: lwf@sdu.edu.cn<sup>b</sup>College of Chemistry, Chemical Engineering and Materials Science, Collaborative Innovation Center of Functionalized Probes for Chemical Imaging in Universities of Shandong, Key Laboratory of Molecular and Nano Probes, Ministry of Education, Shandong Normal University, Jinan, 250014, China. E-mail: yym@sdsnu.edu.cn<sup>c</sup>School of Biological Sciences, Nanyang Technological University, 637551, Singapore. E-mail: ygmu@ntu.edu.sg

† Electronic supplementary information (ESI) available. See DOI: 10.1039/d0na00365d





**Fig. 1** (a) Crystal structures of the BC<sub>3</sub> and C<sub>3</sub>N monolayer. The red rhombus indicates the primitive cell of BC<sub>3</sub>/C<sub>3</sub>N. (b) The sequence of the YAP65WW domain. The positions of strands I–III, connected by hairpins, are indicated above. The initial configurations of the YAP65WW domain on the surface of BC<sub>3</sub> (c) and C<sub>3</sub>N (d) in the simulation box. Carbon, boron and nitrogen atoms are colored in silver, blue and violet, respectively. Na<sup>+</sup> and Cl<sup>−</sup> ions are shown as pink and cyan spheres. Water molecules are not shown for clarity.

the transport of drug species.<sup>33</sup> Thus, element doping to graphene is expected to bring out interesting behaviors at the nano-bio interface compared to the non-polar graphene.

In this study, all-atom molecular dynamics (MD) simulations have been conducted to explore the dynamic and energetic characteristics of a prototypical protein, YAP65WW domain, binding with BC<sub>3</sub> and C<sub>3</sub>N nanosheets. The adsorption processes and structure evolutions of the protein were systematically analyzed and compared. The simulation results revealed that both BC<sub>3</sub> and C<sub>3</sub>N were capable of attacking YAP65WW. It is interesting to find that C<sub>3</sub>N is more capable of inducing the structural unfolding of YAP65WW. This mechanism is because of the binding preference of the protein with the trend from electron-rich nitrogen to electron-deficient boron. We believe that the perspective insights obtained from our study would be beneficial to guide and potentiate the practical applications of graphenic materials and bring about a flourishing new branch in the design of functional bio-nano materials.

## 2. Computational methods

The initial coordinates of YAP65WW were obtained from the crystal structure (PDB code 1JMQ<sup>34</sup> and truncated to include residues 15–40 (ref. 16)). We selected YAP65WW as a model protein because it has been widely used in the study of protein folding kinetics and the interactions with nanomaterials.<sup>35–42</sup> As shown in Fig. 1b, YAP65WW is a signaling and regulatory protein, which exists as a three-stranded, antiparallel  $\beta$ -sheet topology.<sup>43–46</sup> The BC<sub>3</sub> nano-sheet has dimensions of 7.76  $\times$  8.06 nm<sup>2</sup> and is composed of 1620 carbon and 540 boron atoms. Similarly, C<sub>3</sub>N contains 1620 carbon and 540 nitrogen atoms with dimensions of 7.29  $\times$  7.59 nm<sup>2</sup>. The nano-sheet was placed in the *xy* plane and the cross-section of the simulation box was equal to the size of the nano-sheet. Following our previous study,<sup>21</sup> the positions of C<sub>3</sub>N and BC<sub>3</sub> nano-sheets were restrained using a harmonic potential with a force constant of 1000 kJ mol<sup>−1</sup> nm<sup>−2</sup> in all the simulations to mimic a local surface in the whole sample. As depicted in Fig. 1c and d, YAP65WW was then placed above the nano-sheet

with a separation of 2 nm (measured from the center of mass). Then, the complex was solvated in a water box with a height of 7 nm. Periodic boundary conditions were applied on the three directions. For water, the SPC/E water model<sup>47,48</sup> was used. Then, 40 Na<sup>+</sup> and 42 Cl<sup>−</sup> ions were added to neutralize the net charge of YAP65WW and mimic the physiological condition.

The MD simulations were performed using the GROMACS package.<sup>49</sup> The YAP65WW was described by the AMBER99sb force field.<sup>50</sup> For the two nano-sheets, the force field parameters were adopted from our previous work.<sup>18</sup> The long-range electrostatic interactions were treated using the particle mesh Ewald (PME) method.<sup>51,52</sup> The van der Waals (vdW) interactions were calculated with a cutoff distance of 1.2 nm. The covalent bonds involving hydrogen atoms were constrained by the LINCS algorithm.<sup>53</sup> After energy minimization, both systems were equilibrated for 1 ns in the NVT ensemble using the v-rescale thermostat at 300 K,<sup>54</sup> and simulated for another 1 ns in the NPT ensemble with a pressure coupling of 1 bar at the *z*-direction of the simulation box and temperature of 300 K using the Parrinello–Rahman coupling method. During these two stages, position restraints were applied to the heavy atoms of the protein. Then, 1000 ns production simulations were conducted in the NVT ensemble at 300 K. For each nano-sheet, three parallel simulations with different protein orientations and initial velocities were generated for data analysis. In addition to the BC<sub>3</sub> and C<sub>3</sub>N simulations, three control simulations of YAP65WW in water solvent have also been conducted with similar setups to verify the intrinsic stability of protein itself.

## 3. Results and discussions

### 3.1 Quick loading of YAP65WW onto BC<sub>3</sub> and C<sub>3</sub>N

First, we accessed the binding behavior of YAP65WW with two substrates. Initially, YAP65WW was placed above each substrate with a distance of 2 nm. As summarized in Fig. 2, the separation dropped to around 1.2 nm during the first tens of nanoseconds in the simulations (stage one), indicating the quick loading of the protein to the substrates. Then, the separations uniformly



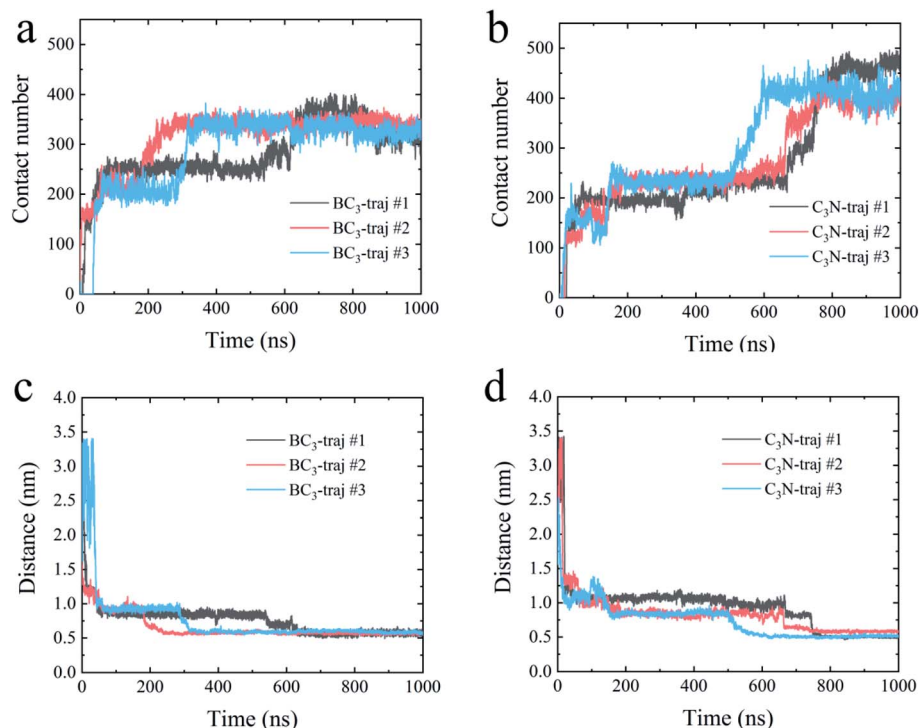


Fig. 2 The contact number between the YAP65WW and BC<sub>3</sub> (a) and C<sub>3</sub>N (b); the separation distance between the center of mass of YAP65WW and BC<sub>3</sub> (c) and C<sub>3</sub>N (d) along the normal directions of the nanosheets.

decreased to around 0.55 to 0.60 nm in three trajectories after hundred nanoseconds of simulations (stage two), although the specific time differed in different trajectories. A similar tendency was also observed when monitoring the number of contacts ( $N_c$ ) between the protein and the substrates. Along with the decrease in distance, YAP65WW established firm contacts with the two substrates. A pair of atoms from the protein and BC<sub>3</sub>/C<sub>3</sub>N, respectively, separated within 0.6 nm was defined as a contacting pair. The  $N_c$  profiles first quickly increased to a plateau of around

80 within 100 ns in two simulations, except that it took a slightly longer time (~150 ns) to reach this plateau in the third trajectory for C<sub>3</sub>N, corresponding to the first stage in the distance analyses. Then, in the second stage, YAP65WW adjusted its conformation until the end of the simulations. It is worth noticing that YAP65WW demonstrated different adsorption behaviors on BC<sub>3</sub> and C<sub>3</sub>N at the end of the simulations. In detail, there are  $124.38 \pm 2.64$  and  $157.90 \pm 4.61$  atomic contacting pairs (each value is averaged over three trajectories) for binding of YAP65WW to BC<sub>3</sub>

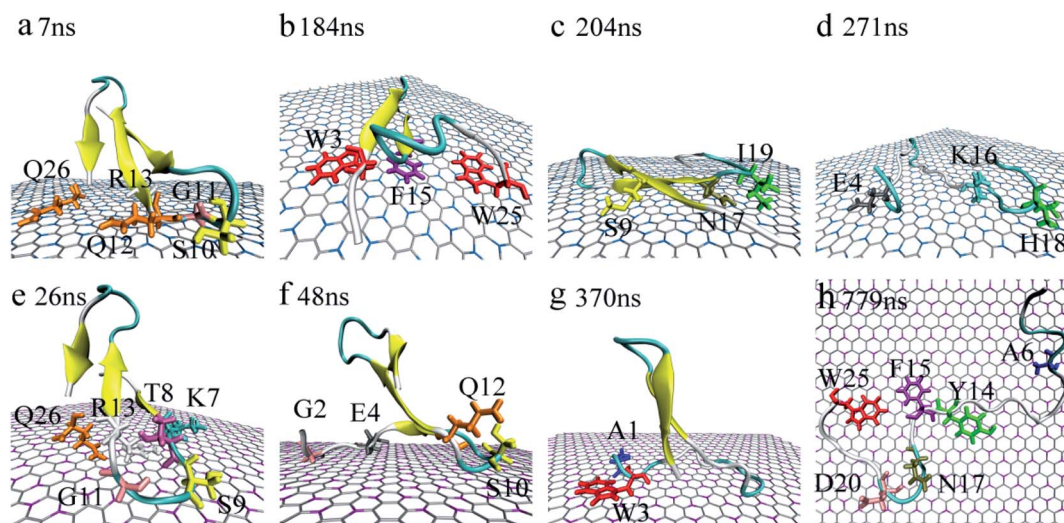


Fig. 3 Structure evolution of the YAP65WW domain during the adsorption process on the surface of BC<sub>3</sub> (a–d) and C<sub>3</sub>N (e–h). Residues that play key roles at the bio-nano interface are highlighted.



and C<sub>3</sub>N, respectively, which accounted for  $57.58 \pm 1.22\%$  and  $73.10 \pm 2.13\%$  of heavy atoms of the protein. From this phenomenon, it is expected that C<sub>3</sub>N is energetically more attractive to YAP65WW because more intimate bindings are established.

### 3.2 Structure disruption of YAP65WW and key residues at the interface

Accompanied by the binding, obvious structure changes were observed for YAP65WW on two nano-sheets. Fig. 3 depicts eight representative snapshots for YAP65WW binding with BC<sub>3</sub> and C<sub>3</sub>N. At the early binding stage, YAP65WW approaches to the BC<sub>3</sub> atoms at  $\sim 7$  ns (Fig. 3a) through five residues, S10, G11, Q12, R13 and Q26 (the contacting time is determined by monitoring the  $N_c$  of each residue with BC<sub>3</sub>/C<sub>3</sub>N, as depicted in Fig. S1 in the ESI†). Then, the protein laid down with its  $\beta$ -sheets stacking parallel to the BC<sub>3</sub> surface (Fig. 3b). Three aromatic key residues, W3, F15 and W25, were found to form intimate  $\pi$ - $\pi$  stacking<sup>55</sup> with BC<sub>3</sub>, which caused a quick flipping over for the strand-3 and more residues (S9, N17 and I19) adsorbed on the BC<sub>3</sub> surface (Fig. 3c). Then, the protein was found to undergo a severe unfolding process (Fig. 3d).

While for C<sub>3</sub>N, YAP65WW contacted to C<sub>3</sub>N at 26 ns by the K7, T8, S9, G11, R13 at the  $\beta$ -hairpin region and Q26 (Fig. 3e). At about 48 ns, the strand-1 of YAP65WW formed firm contacts with C<sub>3</sub>N through G2, E4, S10 and Q12 (Fig. 3f). The unfolding of the  $\beta$ -sheets happened at  $\sim 370$  ns at strand-1 (Fig. 3g). At around 749 ns, YAP65WW completely unfolded (Fig. 3h) with

most residues (A6, Y14, F15, N17, D20 and W25) adsorbed on the C<sub>3</sub>N surface. Despite the different binding processes, it is notable that the  $\pi$ - $\pi$  stacking interactions between the aromatic residues and the substrate appear in both systems. As shown in Fig. S2,† the interfacial water molecules were squeezed once the aromatic residues approached, resulting in nano-scale dewetting. From previous studies, the  $\pi$ - $\pi$  stacking is commonly found in the interactions of graphenic nano-materials with aromatic groups of biomolecules, resulting in severe structural distortions.<sup>15,16,55–58</sup>

We further quantitatively accessed the structural evolution of the protein by calculating the root mean squared deviation (RMSD) of YAP65WW with respect to the crystal structure (only heavy atoms were used in the calculations). As summarized in Fig. 4, the RMSD profiles first increased to around 0.5 nm, particularly for YAP65WW with C<sub>3</sub>N, revealing a slight conformational change from the native structure at the early binding stage. For YAP65WW binding to BC<sub>3</sub>, the totally unfolded state has an RMSD value of around 0.75 nm (Fig. 4a). In contrast, for the case of YAP65WW on C<sub>3</sub>N, RMSD reached 0.8, 1.0 and 1.75 in three trajectories, indicating that the protein structure change is more severe (Fig. 4b). In the control simulations of an isolate YAP65WW in water, the protein maintained a highly ordered native structure with no obvious unfolding being observed (Fig. S3†). This confirms that the unfolding of YAP65WW in the BC<sub>3</sub> and C<sub>3</sub>N simulations is mainly caused by the nano-surface.

It is well known that the spatial structure of protein is mainly maintained by the intra-protein hydrogen bonds (H-bonds).

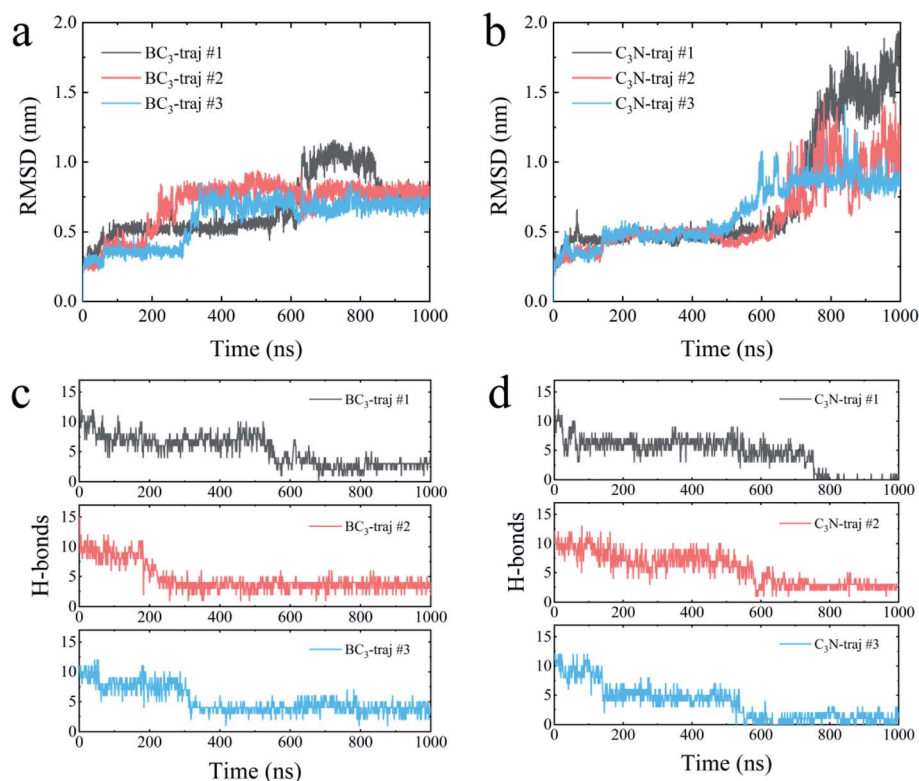


Fig. 4 Time evolutions of the root mean square deviation (RMSD) of the YAP65WW heavy atom relative to the crystal structure upon binding to BC<sub>3</sub> (a) and C<sub>3</sub>N (b). Time evolution of number of intra-protein H-bonds upon binding to BC<sub>3</sub> (c) and C<sub>3</sub>N (d).



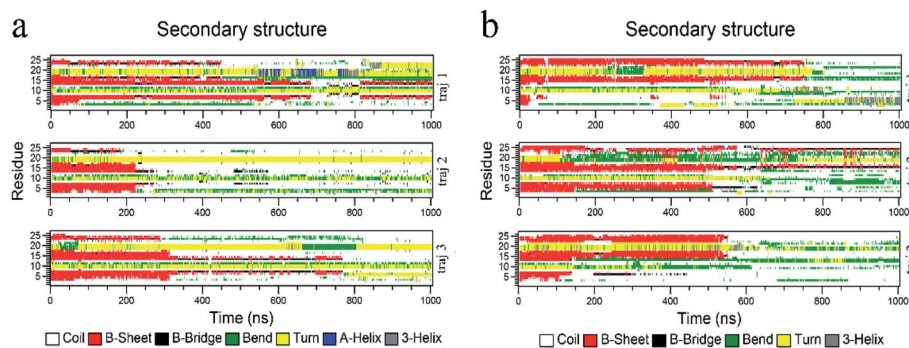


Fig. 5 Time evolution of the secondary structure of YAP65WW upon binding to BC<sub>3</sub> (a) and C<sub>3</sub>N (b). The calculations of the secondary structures were conducted by the `do_dssp` module implemented in the GROMACS package.

During the unfolding process, a clear decrease in the number of H-bonds was observed. As shown in Fig. 4c, the H-bonds decreased from 17 for native YAP65WW to  $2.89 \pm 0.56$ ,  $3.57 \pm 0.78$  and  $3.77 \pm 0.99$  after 1000 ns simulations for YAP65WW binding to BC<sub>3</sub>. However, for C<sub>3</sub>N (Fig. 4d), relatively less H-bonds were maintained and the corresponding values were  $0.10 \pm 0.30$ ,  $2.71 \pm 0.56$  and  $0.95 \pm 0.83$ . This phenomenon confirms that C<sub>3</sub>N causes more severe YAP65WW disruptions than BC<sub>3</sub>. It is also worth mentioning that, although BC<sub>3</sub> induced a smaller ratio of structure loss in YAP65WW, the beginning of YAP65WW unfolding on the BC<sub>3</sub> surface was earlier than that on C<sub>3</sub>N.

### 3.3 Secondary structure evolution of YAP65WW

The biological functions of proteins mainly depend on the secondary structure. Considering this, the in-depth secondary structure analyses of YAP65WW were further conducted using

the define secondary structure of protein (DSSP)<sup>59</sup> program, as summarized in Fig. 5. At the native conformation, YAP65WW contains three antiparallel  $\beta$ -sheet segments:  $\beta$ -sheet 1 (residues 2–8),  $\beta$ -sheet 2 (residues 12–18), and  $\beta$ -sheet 3 (residues 21–25). These three segments were initially preserved during the first 200–300 nanosecond simulations, after which the  $\beta$ -sheets almost disappeared except that small segments showed a dynamic manner of unfolding and folding in the first trajectory of the YAP65WW–BC<sub>3</sub> system and the second trajectory of the YAP65WW–C<sub>3</sub>N system. At the end of all the trajectories, no obvious segments with ordered secondary structures were maintained.

### 3.4 Transverse diffusion of YAP65WW on BC<sub>3</sub> and C<sub>3</sub>N

From the above analyses, both BC<sub>3</sub> and C<sub>3</sub>N are found to be rather attractive to protein because the YAP65WW binding was

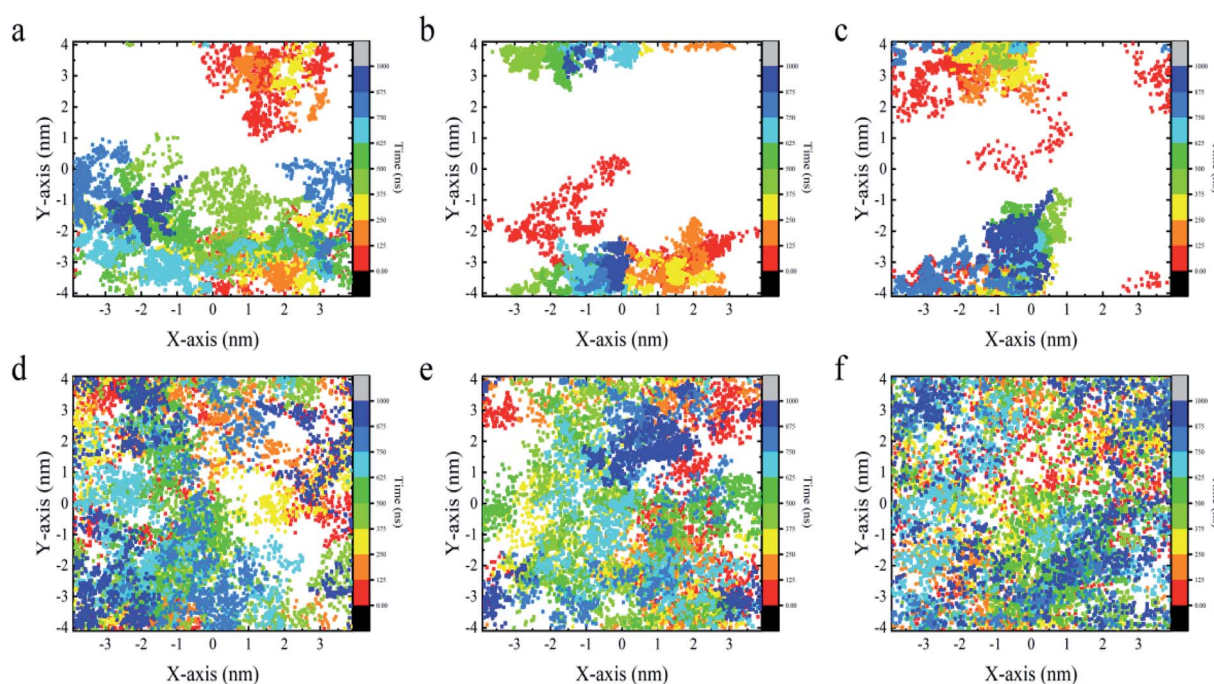


Fig. 6 Transverse diffusion of YAP65WW on the surface of BC<sub>3</sub> (a–c) and C<sub>3</sub>N (d–f) from three parallel trajectories for each substrate. Each dot in the figure depicts the position of the center of mass of protein at certain time.



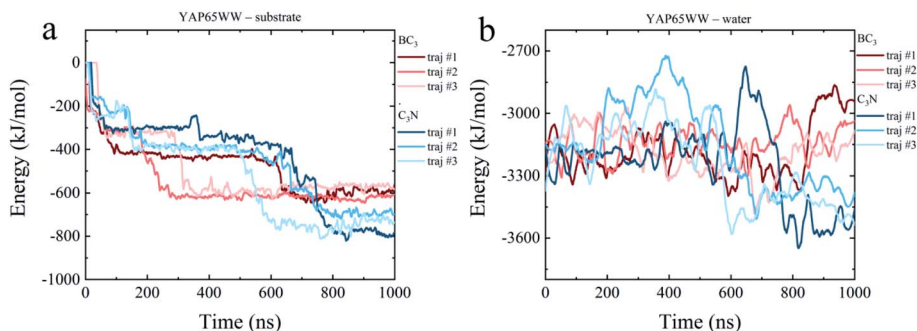


Fig. 7 Time evolutions of interaction energy terms of (a) YAP65WW – substrate and (b) YAP65WW – water.

found to be constantly preserved throughout the simulations. However, differences were also observed between the two substrates. In Fig. 6, we plotted the transverse diffusion of YAP65WW on the two surfaces, which revealed distinct behaviors. For  $BC_3$ , the diffusion of protein is relatively slow, as shown in Fig. 6a–c. During the 1000 ns simulations, the protein only travelled to certain regions of the  $BC_3$  surface. In contrast, during the 1000 ns simulations, the trajectories of YAP65WW almost covered the  $C_3N$  surface, as shown in Fig. 6d–f. This reveals that the transverse diffusion of protein on the  $BC_3$  surface is, to a certain extent, restricted. To further understand the reasons for the different diffusivities, we quantitatively analyzed the bio-nano interface. In Fig. S4 in the ESI,<sup>†</sup> we plotted the distributions of surface water on  $BC_3$  and  $C_3N$ , respectively. It was found that water molecules on  $BC_3$  form a highly ordered structure, which is compatible with the crystal pattern of  $BC_3$ . On the top of hexagonal carbon rings, obvious water clusters are found with high densities. When protein binds with  $BC_3$ , these localized water clusters will definitely hinder the protein diffusion and protect protein from being denatured.

### 3.5 Energetic characteristics of YAP65WW binding to $BC_3$ and $C_3N$

To dissect the origins of the driving force for the YAP65WW and  $BC_3/C_3N$  binding, we have analyzed the interaction energy components from: (i) YAP65WW interacts with substrate and (ii) YAP65WW interacts with water. As shown in Fig. 7a, the adsorptions of YAP65WW to  $BC_3$  and  $C_3N$  are both exergonic. The YAP65WW– $C_3N$  binding releases more energy (averagely  $752.20 \text{ kJ mol}^{-1}$ ) than the YAP65WW– $BC_3$  (averagely  $589.15 \text{ kJ mol}^{-1}$ ), indicating more intimate binding of the former. Particularly, for YAP65WW– $C_3N$  from 500–800 ns, sharp decreases can be observed for the energy profiles that are consistent with the  $N_c$  analyses in Fig. 2a.

The energy term representing the direct interactions of YAP65WW and substrate drives the binding process. In general, the energy released from binding partially compensates the increase in intra-protein potential energy that maintains the folded state. In addition, the interaction energy of the protein with water molecules (abbreviated as  $E_{\text{hydra}}$ ) also changes because of the conformational unfolding upon adsorption. As

summarized in Fig. 7b, for the two systems, the time evolutions of two  $E_{\text{hydra}}$  are found to be quite different. Particularly, it is interesting to find a clear energy decrease for YAP65WW – water in the  $C_3N$  system at 400–700 ns. We attribute this phenomenon to the more severe unfolding of YAP65WW on  $C_3N$  than that on  $BC_3$ , as revealed by the RMSD in Fig. 4a and b. This resulted in more residues exposed to water, which lower the interaction energy to water solvent. This is further supported by monitoring the contacting water around YAP65WW.

To provide a deep understanding of the atomic contribution to the overall binding energy, we plotted the dependence of interaction energy between YAP65WW and substrate with respect to number of contacts ( $N_c$ ) between them (only heavy atoms are considered). As summarized in Fig. 8, a clear linear dependence of the binding strength on  $N_c$  was observed. Second, the larger slopes of the  $C_3N$ –protein represent a stronger binding strength than  $BC_3$ –protein. This reflects that  $C_3N$  is more attractive towards protein, which is consistent to Fig. 7a. Further decomposition of total interaction energy to the elemental level (Fig. S5<sup>†</sup>) indicates a binding preference with the trend from nitrogen boron. Based on these observations, it is believed that elemental doping can efficiently regulate the interactions of graphenic materials with biomacromolecules, which deserve further experimental efforts.

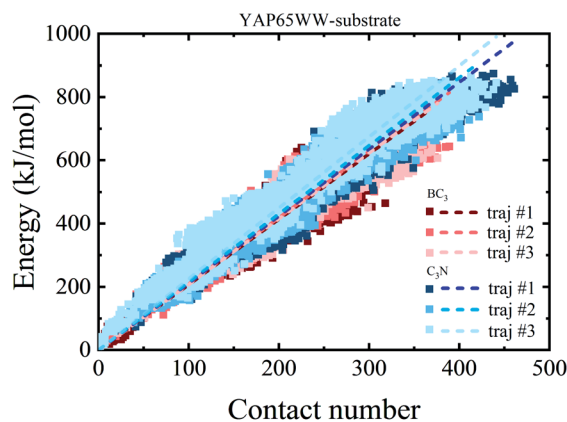


Fig. 8 Relationship between the interaction energy and number of contacts between YAP65WW and substrate (the dash lines indicate linear fit of all data from whole trajectories).



## 4. Conclusion

Using MD simulations, the characteristics of the binding process and structural evolution of protein with graphenic layered materials, B<sub>3</sub>C and C<sub>3</sub>N, were systematically studied. The effects of the elemental doping of boron and nitrogen atoms were compared. Our data clearly revealed that both BC<sub>3</sub> and C<sub>3</sub>N are rather attractive to protein. The binding further induced distortions in protein secondary structures. More importantly, C<sub>3</sub>N demonstrated a stronger capability to destroy the protein structure than BC<sub>3</sub>. In terms of interaction energy, the binding preference of protein follows the trend from electron-rich nitrogen to electron-deficient boron. Localized high density water clusters are formed on the B<sub>3</sub>C surface, which can effectively hinder the transverse diffusion of the protein on it. These findings could be useful to guide the design and fabrication of novel layered nanomaterials with fruitful and tunable biological effects through the elemental doping technique.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

This work is supported by the National Natural Science Foundation of China (Grant no. 11874238), the Basic Research Project of Natural Science Foundation of Shandong Province (Grant no. ZR2018MA034) and Fundamental Research Funds of Shandong University.

## References

- 1 A. K. Geim, *Science*, 2009, **324**, 1530–1534.
- 2 F. Schwierz, *Nat. Nanotechnol.*, 2010, **5**, 487–496.
- 3 U. Kürüm, O. Ö. Ekiz, H. G. Yaglioglu, A. Elmali, M. Ürel, H. Güner, A. K. Mizrak, B. Ortaç and A. Dâna, *Appl. Phys. Lett.*, 2011, **98**, 141103.
- 4 K. V. Sreekanth, S. Zeng, J. Shang, K. T. Yong and T. Yu, *Sci. Rep.*, 2012, **2**, 737–743.
- 5 V. C. Sanchez, A. Jachak, R. H. Hurt and A. B. Kane, *Chem. Res. Toxicol.*, 2012, **25**, 15–34.
- 6 K. Yang, S. Zhang, G. Zhang, X. Sun, S. T. Lee and Z. Liu, *Nano Lett.*, 2010, **10**, 3318–3323.
- 7 C. Li, J. Adamcik and R. Mezzenga, *Nat. Nanotechnol.*, 2012, **7**, 421–427.
- 8 J. Wilson, L. Sloman, Z. He and A. Aksimentiev, *Adv. Funct. Mater.*, 2016, **26**, 4830–4838.
- 9 S. M. Avdoshenko, D. Nozaki, C. Gomes da Rocha, J. W. Gonzalez, M. H. Lee, R. Gutierrez and G. Cuniberti, *Nano Lett.*, 2013, **13**, 1969–1976.
- 10 J. Mao, R. Guo and L. T. Yan, *Biomaterials*, 2014, **35**, 6069–6077.
- 11 R. Guo, J. Mao and L. T. Yan, *Biomaterials*, 2013, **34**, 4296–4301.
- 12 G. Zuo, S. G. Kang, P. Xiu, Y. Zhao and R. Zhou, *Small*, 2013, **9**, 1546–1556.
- 13 Y. Chong, C. Ge, Z. Yang, J. A. Garate, Z. Gu, J. K. Weber, J. Liu and R. Zhou, *ACS Nano*, 2015, **9**, 5713–5724.
- 14 B. Luan, T. Huynh, L. Zhao and R. Zhou, *ACS Nano*, 2015, **9**, 663–669.
- 15 Z. Gu, W. Song, S. H. Chen, B. Li, W. Li and R. Zhou, *Nanoscale*, 2019, **11**, 19362–19369.
- 16 B. Li, D. R. Bell, Z. Gu, W. Li and R. Zhou, *Carbon*, 2019, **146**, 257–264.
- 17 W. Zhang, T. Huynh, P. Xiu, B. Zhou, C. Ye, B. Luan and R. Zhou, *Carbon*, 2015, **94**, 895–902.
- 18 Y. Deng, F. Wang, Y. Liu, Y. Yang, Y. Qu, M. Zhao, Y. Mu and W. Li, *Nanoscale*, 2020, **12**, 5217–5226.
- 19 Y. Liu, X. Song, Y. Yang, Y.-Q. Li, M. Zhao, Y. Mu and W. Li, *Nanoscale*, 2020, **12**, 5209–5216.
- 20 G. Duan, S. G. Kang, X. Tian, J. A. Garate, L. Zhao, C. Ge and R. Zhou, *Nanoscale*, 2015, **7**, 15214–15224.
- 21 B. Li, W. Li, J. M. Perez-Aguilar and R. Zhou, *Small*, 2017, **13**, 1603685.
- 22 Z. Gu, L. Zhao, S. Liu, G. Duan, J. M. Perez-Aguilar, J. Luo, W. Li and R. Zhou, *ACS Nano*, 2017, **11**, 3198–3206.
- 23 T. C. King, P. D. Matthews, H. Glass, J. A. Cormack, J. P. Holgado, M. Leskes, J. M. Griffin, O. A. Scherman, P. D. Barker, C. P. Grey, S. E. Dutton, R. M. Lambert, G. Tustin, A. Alavi and D. S. Wright, *Angew. Chem., Int. Ed. Engl.*, 2015, **54**, 5919–5923.
- 24 J. Mahmood, *Proc. Natl. Acad. Sci. U. S. A.*, 2016, **113**, 6.
- 25 Y. Liu, V. I. Artyukhov, M. Liu, A. R. Harutyunyan and B. I. Yakobson, *J. Phys. Chem. Lett.*, 2013, **4**, 1737–1742.
- 26 Y. Qie, J. Liu, S. Wang, S. Gong and Q. Sun, *Carbon*, 2018, **129**, 38–44.
- 27 J. Xu, J. Mahmood, Y. Dou, S. Dou, F. Li, L. Dai and J. B. Baek, *Adv. Mater.*, 2017, **29**, 1702007.
- 28 M. S. Mahabal, M. D. Deshpande, T. Hussain and R. Ahuja, *ChemPhysChem*, 2015, **16**, 3511–3517.
- 29 S. Mehdi Aghaei, M. M. Monshi, I. Torres, S. M. J. Zeidi and I. Calizo, *Appl. Surf. Sci.*, 2018, **427**, 326–333.
- 30 Y. Tang, X. Cui, W. Chen, D. Zhu, H. Chai and X. Dai, *Appl. Phys. A: Mater. Sci. Process.*, 2018, **124**, 1–13.
- 31 H. Cui, K. Zheng, Y. Zhang, H. Ye and X. Chen, *IEEE Electron Device Lett.*, 2018, **39**, 284–287.
- 32 N. Prasetyo and T. S. Hofer, *J. Chem. Theory Comput.*, 2018, **14**, 6472–6483.
- 33 E. Chigo-Anota, M. A. Alejandro, A. B. Hernández, J. J. S. Torres and M. Castro, *RSC Adv.*, 2016, **6**, 20409–20421.
- 34 M. J. Macias, M. Hyvonen, E. Baraldi, J. Schultz, M. Sudol, M. Saraste and H. Oschkinat, *Nature*, 1996, **382**, 646–649.
- 35 Z. Gu, P. De Luna, Z. Yang and R. Zhou, *Phys. Chem. Chem. Phys.*, 2017, **19**, 3039–3045.
- 36 G. T. Ibragimova and R. C. Wade, *Biophys. J.*, 1999, **77**, 2191–2198.
- 37 M. Jager, H. Nguyen, M. Dendle, M. Gruebele and J. W. Kelly, *Protein Sci.*, 2007, **16**, 1495–1501.
- 38 H. Hu, J. Columbus, Y. Zhang, D. Wu, L. Lian, S. Yang, J. Goodwin, C. Luczak, M. Carter, L. Chen, M. James,



- R. Davis, M. Sudol, J. Rodwell and J. J. Herrero, *Proteomics*, 2004, **4**, 643–655.
- 39 F. Cecconi, C. Guardiani and R. Livi, *Biophys. J.*, 2006, **91**, 694–704.
- 40 M. Petrovich, A. L. Jonsson, N. Ferguson, V. Daggett and A. R. Fersht, *J. Mol. Biol.*, 2006, **360**, 865–881.
- 41 W. P. Russ, D. M. Lowery, P. Mishra, M. B. Yaffe and R. Ranganathan, *Nature*, 2005, **437**, 579–583.
- 42 J. Ricardo Pires, F. Taha-Nejad, F. Toepert, T. Ast, U. Hoffmüller, J. Schneider-Mergener, R. Kühne, M. J. Macias and H. Oschkinat, *J. Mol. Biol.*, 2001, **314**, 1147–11156.
- 43 B. Andre and J. Y. Springael, *Biochem. Biophys. Res. Commun.*, 1994, **205**, 1201–1205.
- 44 P. Bork and M. Sudol, *Trends Biochem. Sci.*, 1994, **19**, 531–533.
- 45 K. Hofmann and P. Bucher, *FEBS Lett.*, 1995, **358**, 153–157.
- 46 J. R. Pires, F. Taha-Nejad, F. Toepert, T. Ast, U. Hoffmüller, J. Schneider-Mergener, R. Kuhne, M. J. Macias and H. Oschkinat, *J. Mol. Biol.*, 2001, **314**, 1147–1156.
- 47 J. Zielkiewicz, *J. Chem. Phys.*, 2005, **123**, 104501.
- 48 B. Hess and N. F. van der Vegt, *J. Phys. Chem. B*, 2006, **110**, 17616–17626.
- 49 M. J. Abraham, T. Murtola, R. Schulz, S. Páll, J. C. Smith, B. Hess and E. Lindahl, *SoftwareX*, 2015, **1–2**, 19–25.
- 50 K. Lindorff-Larsen, S. Piana, K. Palmo, P. Maragakis, J. L. Klepeis, R. O. Dror and D. E. Shaw, *Proteins*, 2010, **78**, 1950–1958.
- 51 T. Darden, D. York and L. Pedersen, *J. Chem. Phys.*, 1993, **98**, 10089–10092.
- 52 U. Essmann, L. Perera, M. L. Berkowitz, T. Darden, H. Lee and L. G. Pedersen, *J. Chem. Phys.*, 1995, **103**, 8577–8593.
- 53 B. Hess, H. Bekker, H. J. C. Berendsen and J. Fraaije, *J. Comput. Chem.*, 1997, **18**, 1463–1472.
- 54 G. Bussi, D. Donadio and M. Parrinello, *J. Chem. Phys.*, 2007, **126**, 014101.
- 55 Z. Yang, Z. Wang, X. Tian, P. Xiu and R. Zhou, *J. Chem. Phys.*, 2012, **136**, 025103.
- 56 G. Zuo, X. Zhou, Q. Huang, H. Fang and R. Zhou, *J. Phys. Chem. C*, 2011, **115**, 23323–23328.
- 57 S. M. Tomásio and T. R. Walsh, *J. Phys. Chem. C*, 2009, **113**, 8778–8785.
- 58 Y. Chong, C. Ge, Z. Yang, J. A. Garate, Z. Gu, J. K. Weber, J. Liu and R. Zhou, *ACS Nano*, 2015, **6**, 5713–5724.
- 59 W. Kabsch and C. Sander, *Biopolymers*, 1983, **22**, 2577–2637.

