



Cite this: *Nanoscale*, 2019, **11**, 9845

Received 12th March 2019,

Accepted 27th April 2019

DOI: 10.1039/c9nr02172h

rsc.li/nanoscale

## InSe: a two-dimensional semiconductor with superior flexibility†

Qinghua Zhao,<sup>a,b,c</sup> Riccardo Frisenda,<sup>b</sup> Tao Wang<sup>b</sup> \*<sup>a,b</sup> and Andres Castellanos-Gomez<sup>b</sup> \*<sup>c</sup>

Two-dimensional indium selenide (InSe) has attracted extensive attention recently due to its record-high charge carrier mobility and photoresponsivity in the fields of electronics and optoelectronics. Nevertheless, the mechanical properties of this material in the ultra-thin regime have not been investigated yet. Here, we present our efforts to determine the Young's modulus of thin InSe (~1–2 layers to ~34 layers) flakes experimentally by using a buckling-based methodology. We find that the Young's modulus has a value of  $23.1 \pm 5.2$  GPa, one of the lowest values reported to date for crystalline two-dimensional materials. This superior flexibility can be very attractive for different applications, such as strain engineering and flexible electronics.

### Introduction

Two-dimensional (2D) semiconductors have attracted a great deal of interest from the scientific community due to their interesting electrical and optical properties that make them promising candidates for future electronic and optoelectronic applications.<sup>1–3</sup> For example, 2D transition metal dichalcogenides (TMDCs) have demonstrated remarkable performances in field-effect transistors (FETs), photodetectors, solar cells, and logic circuits.<sup>4</sup> Another interesting feature of 2D semiconductors is that they can sustain very large deformations without breaking. This has motivated the application of 2D semiconductors in flexible electronic devices and strain engineering studies.<sup>9–12</sup> The flexible electronic devices based on 2-dimensional semi-

conductors stand out due to their high carrier transport mobility, high specific surface area, high optical transparency, excellent mechanical resilience, and environmental stability.<sup>13</sup> Also, the strain engineering on 2D semiconductors, explored first theoretically and then experimentally,<sup>16,17</sup> allows one for the modification of the band gap of these materials by means of mechanical deformations of their lattices.<sup>18–20</sup> These desirable features indicate that 2D semiconductors are a very promising alternative to traditional flexible organic semiconducting devices. Nevertheless, most of the 2D semiconductors studied so far are extremely stiff, having a Young's modulus in the range of ~100–300 GPa (a value much larger than that of organic semiconducting materials (0.1–10 GPa),<sup>23</sup> see Table 1 and Fig. 3a). This means that a high stress is required to deform them. Therefore, flexible (or compliant) 2D semiconductors (characterized by a low Young's modulus) would be desirable for certain applications including (1) flexible electronic applications, where a semiconductor material with a low Young's modulus mismatch with the polymeric/foil substrate is required,<sup>25</sup> (2) force sensing, (3) strain engineering, in which one wants to avoid slippage during strain cycles,<sup>27,28</sup> and (4) nano-electro-mechanical systems, where one wants the resonators to behave in the membrane-like regime.<sup>30,31</sup>

Here, we use the buckling-based metrology method to measure the Young's modulus of thin InSe,<sup>24</sup> a promising 2D semiconductor that shows record-high electron mobility and ultrahigh photoresponse.<sup>33–35</sup> We determine the Young's modulus value  $E = 23.1 \pm 5.2$  GPa and we compare this value with the reported values for the other 2D semiconductors, finding that thin InSe is one order of magnitude more flexible than TMDCs. Our results are relevant for the application of InSe in future flexible electronics, straintronic devices, and nanomechanical systems.

### Results and discussion

2D thin InSe flakes were isolated by mechanical exfoliation from high quality single crystalline InSe ingots grown by the

<sup>a</sup>State Key Laboratory of Solidification Processing, Northwestern Polytechnical University, Xi'an, 710072, P. R. China. E-mail: taowang@nwpu.edu.cn

<sup>b</sup>Key Laboratory of Radiation Detection Materials and Devices, Ministry of Industry and Information Technology, Xi'an, 710072, P. R. China

<sup>c</sup>Materials Science Factory, Instituto de Ciencia de Materiales de Madrid (ICMM-CSIC), Madrid, E-28049, Spain. E-mail: andres.castellanos@csic.es, riccardo.frisenda@csic.es

† Electronic supplementary information (ESI) available: Characterization of bulk InSe (SEM, EDX, XRD and TEM), details on the AFM thickness determination and on the finite element analysis simulation. See DOI: 10.1039/c9nr02172h



**Table 1** The comparison of the Young's modulus of two-dimensional (2D) materials measured at room temperature and an ambient environment. Legend: "Isolation method"; ME: mechanical exfoliation, CVD: chemical vapor deposition, VDWE: van der Waals epitaxy. "Testing method"; (1) spring constant scaling, (2) nano-indentation, (3) compliance maps, (4) electrostatic deflection, (5) blister test, (6) constant force maps, (7) nano-resonator, (8) bimodal AFM, (9) buckling metrology method, and (10) micro-tensile method

Type	Materials	Isolation method	# of layers/ thickness	Young's modulus (GPa) (testing method)		Ref.
				Highest	Lowest	
Semi-metal	InSe	ME	~1–2L to 34L	23.1 ± 5.2 (9)		This work
	Graphene	ME	1–100L	1000 ± 100 (2)	430 (1)	5 and 6
Semiconductor	Graphene	CVD	1L	1000 ± 50 (2)	160 (2)	7 and 8
		MoS <sub>2</sub>	ME	1–108L	330 ± 70 (2)	200 (7)
	MoSe <sub>2</sub>	CVD	1–2L	264 ± 18 (2)	197 ± 31 (5)	21 and 22
		ME	5–10L	224 ± 41 (9)		24
	WS <sub>2</sub>	CVD	1–2L	177.2 ± 9.3 (10)		26
		ME	3–8L	236 ± 65 (9)		24
	WSe <sub>2</sub>	CVD	1L	272 ± 18 (2)		21
		ME	4–12L	167 ± 7 (2)	163 ± 39 (9)	24 and 29
	BP	ME	15–25 nm	Zz: 58.6 ± 11.7, Ac: 27.2 ± 4.1 (2)		32
	GaS	ME	~5–20 nm	173 ± 15 (2)	~50 (2)	36
	GaSe	ME	~10–30 nm	82 ± 13 (2)		36
	GaTe	ME	~10–30 nm	~75 (2)	25 ± 16 (2)	36
Insulator	h-BN	CVD	1 nm, 15 nm	1160 ± 100 (2)	223 ± 16 (2)	41 and 42
	Graphene oxide	—	1L	207.6 ± 23.4 (6)		43
		Reduced	1L	250 ± 150 (1)		44
	Mica	ME	2–14L	200 ± 30 (2)	170 ± 40 (1)	46
	Na <sub>0.5</sub> -Fluo	ME	12–90L	21 ± 9 (1)		47
	Vermiculite	ME	>2L	175 ± 16 (6)		48
Topological insulator	Bi <sub>2</sub> Te <sub>3</sub>	VDWE	5–14 nm	11.7–25.7 (2)		49
	Bi <sub>2</sub> Se <sub>3</sub>	VDWE	7–12L	17.48–25.45 (2)		52

Bridgman method.<sup>37</sup> Bulk InSe have been characterized by scanning electron microscopy (SEM), X-ray diffraction (XRD) and transmission electronic microscopy (TEM) to ensure its high crystallinity and to identify its polytope. More details about the characterization can be found in section 1 of the ESI.† We address the reader to the Materials and methods section for details about the InSe exfoliation. In order to measure the mechanical properties of InSe flakes through the buckling metrology method, they have to be deposited onto a very compliant substrate and subjected to uniaxial compression. In the case of InSe deposited on top of Gel-Film (by Gel-Pak®), if the compression strain in the InSe is above a critical value of approximately 0.07%, the flakes undertake a buckling instability.<sup>38</sup> Due to the competition between the buckling of the flakes and the adhesion with the substrate, a wavy pattern (ripples) appears on the flakes. Interestingly, the period of these ripples ( $\lambda$ ) depends only on the mechanical properties of the flake and substrate:<sup>38–40</sup>

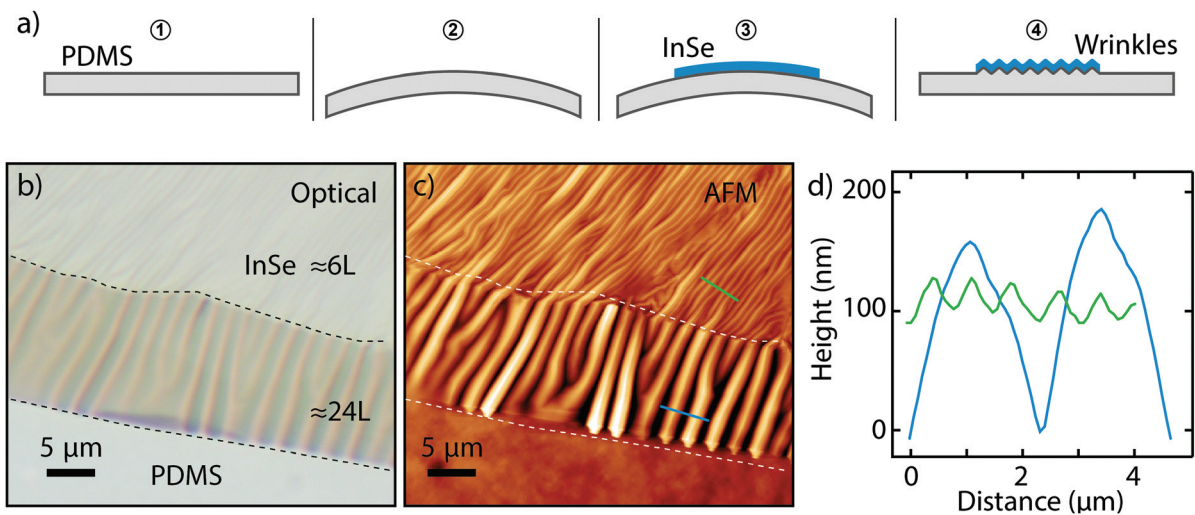
$$\lambda = h \left[ \frac{8\pi^3(1 - \nu_s^2)E_f}{3(1 - \nu_f^2)E_s} \right]^{\frac{1}{3}} \quad (1)$$

where  $h$  is the thickness of the flake,  $\nu_s$  and  $\nu_f$  correspond to the Poisson's ratio of the substrate and flake and  $E_s$  and  $E_f$  are the Young's modulus of the substrate and flake, respectively. Therefore, for a known substrate one can extract the Young's modulus of InSe by measuring the period of ripples for flakes with different thicknesses. We have recently demonstrated the suitability of this method to probe the mechanical properties

of 2D materials and we address the readers to ref. 24 for more details on the technique.

Fig. 1a shows a cartoon of the fabrication process followed to apply the uniaxial compression to InSe flakes. A rectangular shape Gel-Film substrate is slightly bent to induce a uniaxial expansion of its topmost surface. Then InSe flakes are transferred on the bent surface by mechanical exfoliation with Nitto tape (see Materials and methods) and the substrate stress is released yielding the desired uniaxial compression. Fig. 1b shows a transmission mode optical microscopy image of an isolated InSe flake fabricated following this approach. The flake presents a marked wavy pattern because of the buckling instability induced ripples. The corresponding atomic force microscopy (AFM) topography image is shown in Fig. 1b with the thickness ranging from ~6L to ~24L (9.4 nm and 24 nm thick). The height calculated from the AFM dynamic mode measurements presented in this work has a systematic offset that we determined to be  $4.9 \pm 0.5$  nm in the case of InSe (this value should be subtracted from the AFM data in order to find the real flake thickness, see section 2 of the ESI†). This offset, which has been reported also for other 2D materials, is caused by the different interaction between the tip and either the substrate or the flake.<sup>45</sup> It is important to note that this offset does not influence the determination of the Young's modulus of InSe and thus we did not subtract it from the data presented in this article. Fig. 1c shows the sinusoidal line profile shape of the ripples. From Fig. 1b and c, one can see that the wavelength  $\lambda$  of the ripples is strongly thickness dependent.

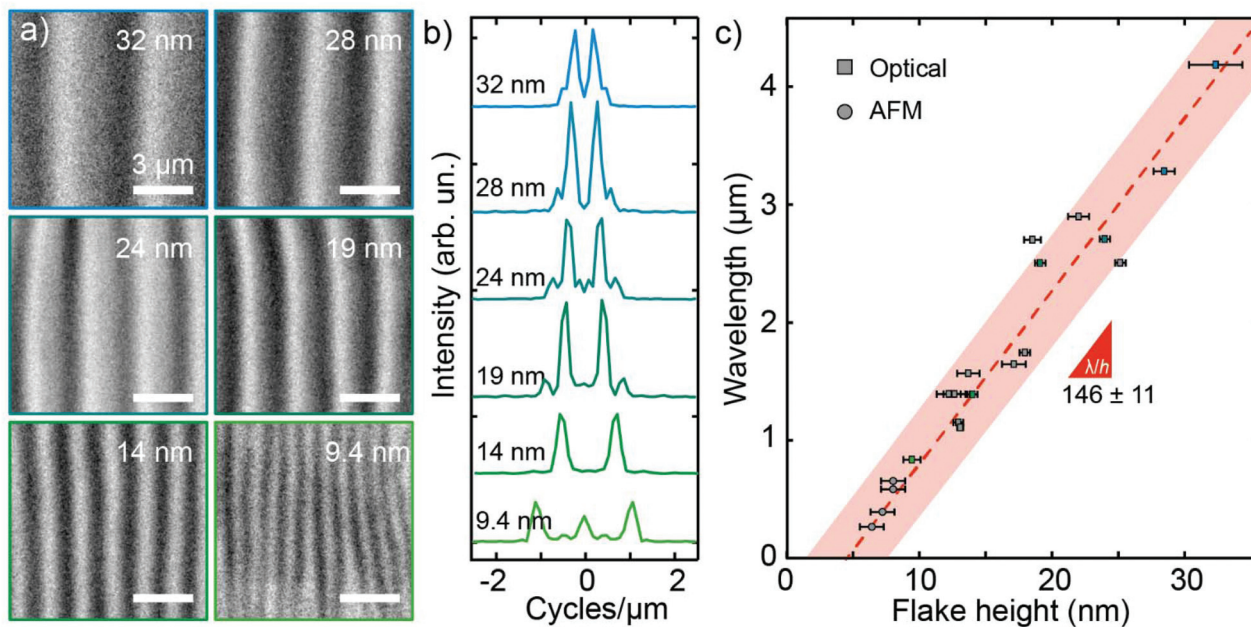




**Fig. 1** The wrinkled thin InSe flakes. (a) The schematic diagram of fabricating wrinkled InSe flakes on the Gel-Film substrate by using the buckling metrology method. (b, c) Transmission mode optical pictures (b) and surface morphology recorded by AFM (c) of a wrinkled InSe flake with thicknesses ( $h$ ) of  $\sim 6L$  and  $\sim 24L$ . (d) The line profiles of the wrinkles with different thicknesses recorded at the positions marked in panel c (green line, 6L; blue line, 24L).

As seen in eqn (1), the Young's modulus can be determined by measuring the wavelength of the buckling induced ripples for flakes of different thicknesses. Fig. 2a shows six grayscale transmission mode optical images of the ripples produced on InSe flakes with different thicknesses. The wavelength of the ripples can be accurately measured through the fast Fourier transform (FFT) of these images (Fig. 2b). Fig. 2c shows a summary of the wavelength values measured from 20 InSe

flakes with thicknesses ranging from 6.4 nm to 32 nm ( $\sim 1$ –2 layers to 34 layers). Note that rippled patterns with wavelengths smaller than  $\sim 1 \mu\text{m}$  and amplitude  $< 10$  nm cannot be well-resolved with optical microscopy.<sup>24</sup> The experimental data points follow a marked linear trend, as expected from eqn (1). From the slope  $\lambda/h = 146 \pm 11$  we can determine the Young's modulus of InSe. Using the known values of the Poisson's ratio of PDMS (Gel-Film)  $\nu_s = 0.5$  and InSe flake  $\nu_f = 0.27$ ,<sup>50,51</sup>



**Fig. 2** The determination of Young's modulus of thin InSe flakes. (a) Grayscale transmission mode optical microscopy images of wrinkled patterns on InSe flakes with thicknesses ranging from 9.4 nm to 32 nm. (b) The line cuts along the FFT maxima extracted from the optical images in panel a. (c) Relationship between the wavelength of the wrinkles and the thickness of the InSe flakes. The red dashed line represents the linear fit based on the data and the light red shaded area indicates the uncertainty of the fitting.



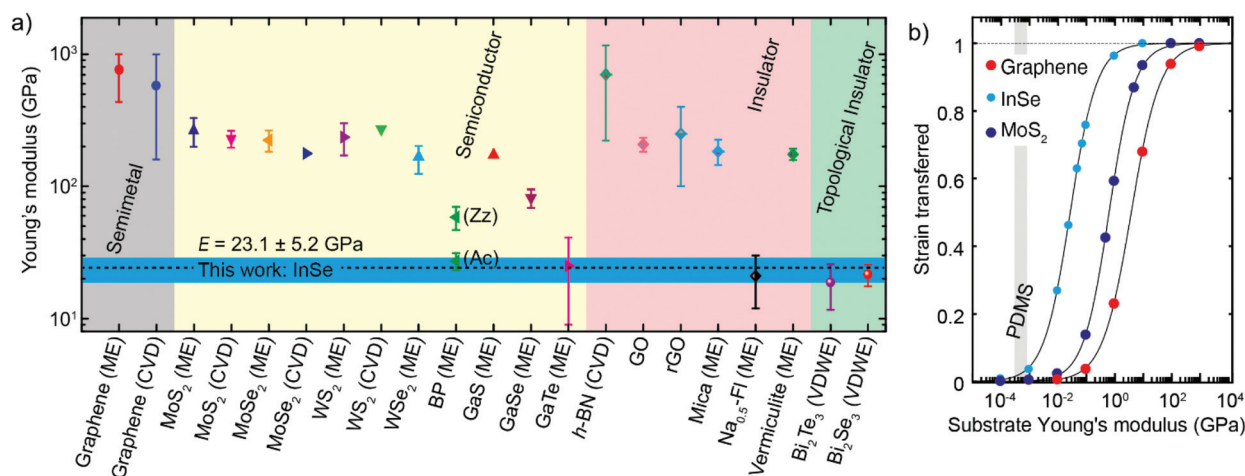


and the measured Young's modulus of our Gel-Film  $E_s = 492 \pm 11$  kPa (see the ESI of ref. 24),<sup>24</sup> we determine the Young's modulus of thin InSe flakes,  $E = 23.1 \pm 5.2$  GPa, a value much smaller than the reported Young's modulus of other 2D materials.

In order to put this value in the more general context of 2D materials, we show in Fig. 3a a graphical comparison between the Young's modulus values for various isolated 2D materials available in the literature in a semi-logarithm scale plot. We indicate the highest and lowest Young's modulus reported in the literature through the error bar. The four differently colored regions correspond to semimetal, semiconductor, insulator and topological insulator groups, respectively. The value determined with the buckling method of thin InSe Young's modulus is shown with a black dashed line and a surrounding blue shadow (that represents the uncertainty of our measurement result). According to the plot, the thin InSe flakes have a Young's modulus value which is around two orders of magnitude smaller than that of graphene and one order of magnitude lower than MoS<sub>2</sub> flakes. This value of the Young's modulus is among the lowest values reported for 2D materials to date and is comparable only to that of metal-organic frameworks (MOFs,  $\sim 5$  GPa).<sup>53</sup> Notably, this small value for the Young's modulus of InSe is in agreement with theoretical calculations, which predict an isotropic Young's modulus for monolayer InSe of 57 GPa and that this value is much smaller than the predicted values for MoS<sub>2</sub> and graphene.<sup>50</sup> All the details displayed in Fig. 3a are also summarized in Table 1 to facilitate a quantitative comparison between different materials.

The low Young's modulus value of InSe has implications in its applicability in flexible electronics, strain engineering or sensors. For example, applying or transferring strain to a 2D material with a low Young's modulus requires less force than

transferring the same amount of strain to a stiffer 2D material. To understand the role of the Young's modulus in the transfer of strain to a 2D material deposited on a substrate, we performed a three-dimensional axisymmetric finite element analysis (FEA) using the software COMSOL Multiphysics (version 5.1). The results shown in Fig. 3b are calculated for a thin InSe with a thickness of 10 nm, a diameter of 20  $\mu\text{m}$  and the Young's modulus  $E_{\text{InSe}} = 23.1$  GPa, placed on a substrate with a thickness of 1000  $\mu\text{m}$  and a diameter of 10 000  $\mu\text{m}$ . Additional details about the FEA calculation can be found in section 3 of the ESI.<sup>†</sup> The interface between the InSe flake and the substrate is modeled using perfect bonding. In each step of the simulation we allowed the substrate to expand and extracted the total expansion induced in the InSe flake from which we calculated the strain transfer for all the different substrate Young's modulus values. We repeated the calculation for MoS<sub>2</sub> ( $E_{\text{MoS}_2} = 250$  GPa) and graphene ( $E_{\text{Graphene}} = 1$  TPa).<sup>5,7,24</sup> The calculations show that, independently of the Young's modulus of the 2D material, for very small values of the substrate Young's modulus no strain is transferred from the substrate to the 2D flake, while for very large values the strain transfer approaches 100%. In between these two limits one can see that the strain transfer presents in all three cases a similar sigmoidal shape and is shifted along the horizontal axis. The onset of transfer of each curve depends on the 2D material Young's modulus being the lowest for InSe and the largest for graphene. For 2D flakes with a lower Young's modulus the strain transfer is larger than for flakes with a larger modulus (given the same substrate Young's modulus). Since no atomistic details are taken into account in the simulation, the real absolute value of strain transfer for a given substrate's Young's modulus can differ from the calculated one. Nevertheless, the general shape of the strain transfer curve and the trend observed should hold true for all the different 2D materials.



**Fig. 3** Summary of the experimentally measured Young's modulus values of isolated two-dimensional species (semimetals, semiconductors, insulators and topological insulators). (a) The different values of the Young's modulus of 2D materials reported in the literature are compared with the value of InSe obtained in this work  $E = 23.1 \pm 5.2$  GPa (centered at the black dashed line with the blue shaded area indicating the experimental uncertainty). (b) Strain transfer as a function of the substrate Young's modulus calculated from finite element analysis for InSe, MoS<sub>2</sub> and graphene. The grey shaded regions indicate the Young's modulus values of polymer substrates typically used in flexible and printed electronics.



## Conclusions

In conclusion, we reported the experimental value of the Young's modulus of thin InSe flakes using the buckling metrology method. We find an ultralow value in the range of 18–28 GPa, which makes thin InSe one of the most flexible two-dimensional materials. This low Young's modulus value can prove to be highly valuable in applications such as force sensing (as the same force will produce a larger deformation in InSe than in other 2D semiconductors) or flexible electronic applications (where a small mismatch between the Young's modulus of the substrate and the 2D semiconductor is required to avoid stress accumulation after several flexing cycles). Also, we calculate that such superior flexibility may facilitate more effective strain transfer from a polymeric substrate to InSe flakes deposited on top, making InSe a very interesting material for applications in future straintronic devices.

## Materials and methods

### Materials and sample fabrication

The high-quality InSe crystal was grown by the Bridgman method. With the atomic ratio In:Se = 52:48, high purity indium (99.9999%) and selenium (99.9999%) elements are sealed into a quartz crucible under the high vacuum conditions ( $2 \times 10^{-5}$  Pa). Then the quartz crucible was transferred into a rocking furnace and kept swinging for 48 h at 650 °C to mix the melt completely. Then a vertical Bridgman furnace with a temperature gradient of 10 °C cm<sup>-1</sup> was used for crystal growth. Detailed characterization of the high-quality bulk materials can be found in section 1 of the ESI.† The elastomeric compliant substrate used in this work is commercially available polydimethylsiloxane (PDMS) supplied by Gel-Pak (WF 4x Gel-Film), whose Young's modulus ( $E_s = 492 \pm 11$  kPa) has been determined in our previous work.<sup>24</sup> A mechanical exfoliation method with Scotch tape (3M®) and Nitto tape (Nitto Denko® SPV 224) was used for thin InSe flake fabrication. Then the flakes were transferred onto a curved Gel-Film substrate to induce wrinkles. All the exfoliation processes are carried out in an ambient environment and at room temperature.

### Determination of wrinkle size and flake thickness

The images of rippled InSe flakes onto PDMS were acquired in optical microscopy transmission mode (Motic® BA310 MET-T) in the case of wrinkles with wavelengths larger than  $\sim 1$   $\mu$ m and with dynamic mode atomic force microscopy in the case of wavelengths smaller than 1  $\mu$ m. After that, these rippled flakes were transferred onto SiO<sub>2</sub> (280 nm)/Si substrates for AFM measurements. Gwyddion software was used for the quantitative analysis of the wavelength of the ripples on the flakes. The thickness of InSe flakes was determined by using an ezAFM from Nanomagntics operating in dynamic mode. A Tap190Al-G cantilever by BudgetSensors with a force constant of 40 N m<sup>-1</sup> and a resonance frequency of 300 kHz was used.

To take into account the offset present in the thickness determination of 2D flakes by dynamic mode AFM,<sup>45</sup> one has to subtract a constant value of 4.9 nm from the raw AFM heights. See section S2 of the ESI† for more details on the estimation of this offset. Note that all the thickness values given in the main text include this offset and thus to find the real thickness one has to subtract 4.9 nm from the reported height.

## Conflicts of interest

There are no conflicts to declare.

## Acknowledgements

This project has received funding from the European Research Council (ERC) under the European Union's Horizon 2020 research and innovation program (grant agreement no. 755655, ERC-StG 2017 project 2D-TOPSENSE). EU Graphene Flagship funding (Grant Graphene Core 2, 785219) is acknowledged. RF acknowledges support from the Spanish Ministry of Economy, Industry and Competitiveness through a Juan de la Cierva-formación fellowship (2017 FJCI-2017-32919). QHZ acknowledges the grant from the China Scholarship Council (CSC) under no. 201700290035. TW acknowledges support from the National Natural Science Foundation of China: 51672216.

## References

- Q. H. Wang, K. Kalantar-Zadeh, A. Kis, J. N. Coleman and M. S. Strano, *Nat. Nanotechnol.*, 2012, **7**, 699–712.
- K. F. Mak and J. Shan, *Nat. Photonics*, 2016, **10**, 216.
- J. F. Sánchez-Royo, G. Muñoz-Matutano, M. Brotons-Gisbert, J. P. Martínez-Pastor, A. Segura, A. Cantarero, R. Mata, J. Canet-Ferrer, G. Tobias and E. Canadell, *Nano Res.*, 2014, **7**, 1556–1568.
- S. Manzeli, D. Ovchinnikov, D. Pasquier, O. V. Yazyev and A. Kis, *Nat. Rev. Mater.*, 2017, **2**, 17033.
- C. Lee, X. Wei, J. W. Kysar and J. Hone, *science*, 2008, **321**, 385–388.
- F. Traversi, F. J. Gúzman-Vázquez, L. G. Rizzi, V. Russo, C. S. Casari, C. Gómez-Navarro and R. Sordan, *New J. Phys.*, 2010, **12**, 023034.
- G.-H. Lee, R. C. Cooper, S. J. An, S. Lee, A. Van Der Zande, N. Petrone, A. G. Hammerberg, C. Lee, B. Crawford and W. Oliver, *Science*, 2013, **340**, 1073–1076.
- C. S. Ruiz-Vargas, H. L. Zhuang, P. Y. Huang, A. M. Van Der Zande, S. Garg, P. L. McEuen, D. A. Muller, R. G. Hennig and J. Park, *Nano Lett.*, 2011, **11**, 2259–2263.
- R. Roldán, A. Castellanos-Gomez, E. Cappelluti and F. Guinea, *J. Phys.: Condens. Matter*, 2015, **27**, 313201.
- D. Akinwande, N. Petrone and J. Hone, *Nat. Commun.*, 2014, **5**, 5678.
- S. Deng, A. V. Sumant and V. Berry, *Nano Today*, 2018, **22**, 14–35.



- 12 Z. Dai, L. Liu and Z. Zhang, *Adv. Mater.*, 2019, 1805417.
- 13 S. J. Kim, K. Choi, B. Lee, Y. Kim and B. H. Hong, *Annu. Rev. Mater. Res.*, 2015, **45**, 63–84.
- 14 A. Castellanos-Gomez, M. Poot, G. A. Steele, H. S. van der Zant, N. Agrait and G. Rubio-Bollinger, *Adv. Mater.*, 2012, **24**, 772–775.
- 15 J. Lee, Z. Wang, K. He, J. Shan and P. X.-L. Feng, *ACS Nano*, 2013, **7**, 6086–6091.
- 16 J. Feng, X. Qian, C.-W. Huang and J. Li, *Nat. Photonics*, 2012, **6**, 866.
- 17 H. Li, A. W. Contryman, X. Qian, S. M. Ardakani, Y. Gong, X. Wang, J. M. Weisse, C. H. Lee, J. Zhao and P. M. Ajayan, *Nat. Commun.*, 2015, **6**, 7381.
- 18 M. Brotons-Gisbert, D. Andres-Penares, J. Suh, F. Hidalgo, R. Abargues, P. J. Rodríguez-Cantó, A. Segura, A. Cros, G. Tobias and E. Canadell, *Nano Lett.*, 2016, **16**, 3221–3229.
- 19 Y. Li, T. Wang, H. Wang, Z. Li, Y. Chen, D. West, R. Sankar, R. K. Ulaganathan, F. Chou and C. Wetzel, *Nano Lett.*, 2018, **18**, 5078–5084.
- 20 Y. Li, T. Wang, M. Wu, T. Cao, Y. Chen, R. Sankar, R. K. Ulaganathan, F. Chou, C. Wetzel and C.-Y. Xu, *2D Mater.*, 2018, **5**, 021002.
- 21 K. Liu, Q. Yan, M. Chen, W. Fan, Y. Sun, J. Suh, D. Fu, S. Lee, J. Zhou and S. Tongay, *Nano Lett.*, 2014, **14**, 5097–5103.
- 22 D. Lloyd, X. Liu, N. Boddeti, L. Cantley, R. Long, M. L. Dunn and J. S. Bunch, *Nano Lett.*, 2017, **17**, 5329–5334.
- 23 B. H. Stuart, *Polymer analysis*, John Wiley & Sons, 2008.
- 24 N. Iguñiz, R. Frisenda, R. Bratschitsch and A. Castellanos-Gomez, *Adv. Mater.*, 2019, 1807150.
- 25 H. Gleskova, I.-C. Cheng, S. Wagner, J. C. Sturm and Z. Suo, *Sol. Energy*, 2006, **80**, 687–693.
- 26 Y. Yang, X. Li, M. Wen, E. Hacıoğlu, W. Chen, Y. Gong, J. Zhang, B. Li, W. Zhou and P. M. Ajayan, *Adv. Mater.*, 2017, **29**, 1604201.
- 27 R. Frisenda, M. Drüppel, R. Schmidt, S. M. de Vasconcellos, D. P. de Lara, R. Bratschitsch, M. Rohlfing and A. Castellanos-Gomez, *npj 2D Mater. Appl.*, 2017, **1**, 10.
- 28 C. Rice, R. Young, R. Zan, U. Bangert, D. Wolverson, T. Georgiou, R. Jalil and K. Novoselov, *Phys. Rev. B: Condens. Matter Mater. Phys.*, 2013, **87**, 081307.
- 29 R. Zhang, V. Koutsos and R. Cheung, *Appl. Phys. Lett.*, 2016, **108**, 042104.
- 30 A. Castellanos-Gomez, R. van Leeuwen, M. Buscema, H. S. van der Zant, G. A. Steele and W. J. Venstra, *Adv. Mater.*, 2013, **25**, 6719–6723.
- 31 A. Castellanos-Gomez, V. Singh, H. S. van der Zant and G. A. Steele, *Ann. Phys.*, 2015, **527**, 27–44.
- 32 J. Tao, W. Shen, S. Wu, L. Liu, Z. Feng, C. Wang, C. Hu, P. Yao, H. Zhang and W. Pang, *ACS Nano*, 2015, **9**, 11362–11370.
- 33 D. A. Bandurin, A. V. Tyurnina, L. Y. Geliang, A. Mishchenko, V. Zólyomi, S. V. Morozov, R. K. Kumar, R. V. Gorbachev, Z. R. Kudrynskiy and S. Pezzini, *Nat. Nanotechnol.*, 2017, **12**, 223.
- 34 S. A. Wells, A. Henning, J. T. Gish, V. K. Sangwan, L. J. Lauhon and M. C. Hersam, *Nano Lett.*, 2018, **18**, 7876–7882.
- 35 M. Li, C. Y. Lin, S. H. Yang, Y. M. Chang, J. K. Chang, F. S. Yang, C. Zhong, W. B. Jian, C. H. Lien and C. H. Ho, *Adv. Mater.*, 2018, 1803690.
- 36 B. Chitara and A. Ya'akovovitz, *Nanoscale*, 2018, **10**, 13022–13027.
- 37 T. Ishii, *J. Cryst. Growth*, 1988, **89**, 459–462.
- 38 D. Y. Khang, J. A. Rogers and H. H. Lee, *Adv. Funct. Mater.*, 2009, **19**, 1526–1536.
- 39 C. M. Stafford, C. Harrison, K. L. Beers, A. Karim, E. J. Amis, M. R. VanLandingham, H.-C. Kim, W. Volksen, R. D. Miller and E. E. Simonyi, *Nat. Mater.*, 2004, **3**, 545.
- 40 A. Volynskii, S. Bazhenov, O. Lebedeva and N. Bakeev, *J. Mater. Sci.*, 2000, **35**, 547–554.
- 41 S. M. Kim, A. Hsu, M. H. Park, S. H. Chae, S. J. Yun, J. S. Lee, D.-H. Cho, W. Fang, C. Lee and T. Palacios, *Nat. Commun.*, 2015, **6**, 8662.
- 42 L. Song, L. Ci, H. Lu, P. B. Sorokin, C. Jin, J. Ni, A. G. Kvashnin, D. G. Kvashnin, J. Lou and B. I. Yakobson, *Nano Lett.*, 2010, **10**, 3209–3215.
- 43 J. W. Suk, R. D. Piner, J. An and R. S. Ruoff, *ACS Nano*, 2010, **4**, 6557–6564.
- 44 C. Gómez-Navarro, M. Burghard and K. Kern, *Nano Lett.*, 2008, **8**, 2045–2049.
- 45 P. Nemes-Incze, Z. Osváth, K. Kamarás and L. Biró, *Carbon*, 2008, **46**, 1435–1442.
- 46 A. Castellanos-Gomez, M. Poot, A. Amor-Amorós, G. A. Steele, H. S. van der Zant, N. Agrait and G. Rubio-Bollinger, *Nano Res.*, 2012, **5**, 550–557.
- 47 D. A. Kunz, E. Max, R. Weinkamer, T. Lunkenbein, J. Breu and A. Fery, *Small*, 2009, **5**, 1816–1820.
- 48 J. W. Suk, R. D. Piner, J. An and R. S. Ruoff, *Thin Solid Films*, 2013, **527**, 205–209.
- 49 L. Guo, H. Yan, Q. Moore, M. Buettner, J. Song, L. Li, P. T. Araujo and H.-T. Wang, *Nanoscale*, 2015, **7**, 11915–11921.
- 50 T. Hu, J. Zhou and J. Dong, *Phys. Chem. Chem. Phys.*, 2017, **19**, 21722–21728.
- 51 R. H. Pritchard, P. Lava, D. Debruyne and E. M. Terentjev, *Soft Matter*, 2013, **9**, 6037–6045.
- 52 H. Yan, C. Vajner, M. Kuhlman, L. Guo, L. Li, P. T. Araujo and H.-T. Wang, *Appl. Phys. Lett.*, 2016, **109**, 032103.
- 53 C. Hermosa, B. R. Horrocks, J. I. Martínez, F. Liscio, J. Gómez-Herrero and F. Zamora, *Chem. Sci.*, 2015, **6**, 2553–2558.

