RSC Advances

PAPER

Cite this: RSC Adv., 2018, 8, 35753

Fast and safe synthesis of micron germanium in an ammonia atmosphere using Mo₂N as catalyst

Baojun Ma[,](http://orcid.org/0000-0003-0020-6650) \mathbb{D}^* Dekang Li, Xiaoyan Wang and Keying Lin

Here, we reported a new method for fast and safe synthesis of a micron germanium (Ge) semiconductor. The Ge was successfully prepared from mixed GeO₂ with a low amount of MoO₃ by the NH₃ reduction method at 800 \degree C for an ultra-short time of 10 min. XRD patterns show that the Ge has a tetragonal structure. SEM images show that the size of the Ge particles is from 5 μ m to 10 μ m, and so it is on the micron scale. UV-visible diffuse reflectance spectroscopy shows that the Ge has good light absorption both in the ultraviolet and visible regions. The formation of Ge mainly goes through a two-step conversion in the NH₃ flow. Firstly, GeO₂ is converted to Ge₃N₄, and then Ge₃N₄ is decomposed to generate Ge. The comparison experiments of $MoO₃$ and $Mo₂N$ demonstrate that $Mo₂N$ is the catalyst for the Ge synthesis which improves the Ge_3N_4 decomposition. The presented fast and safe synthesis method of Ge has great potential for industrialization and the proposed Mo₂N boosting the Ge₃N₄ decomposition has provided significant guidance for other nitride decomposition systems. **PAPER**
 EXERCISE SO FRAME SO TRICTON DETERMINDENT CONDUCTS CREAT AND STATE SO FRAME SO FRAME IN A SECTION CHEME CHEME AND SERVICE AND SERVICE SO FROM A SCHOOL AND SERVICE SO FROM A SCHOOL AND SOLUTION (SO PARTICULAR CRE

Received 3rd September 2018 Accepted 7th October 2018

DOI: 10.1039/c8ra07352j

rsc.li/rsc-advances

1 Introduction

With the development of advanced electronic information industries, semiconductor materials have attracted widespread attention due to their unique optical and electrical properties.¹–⁶ Ge as an important semiconductor material with stable chemical properties and obvious non-metallic properties has many advantages, such as non-toxicity, biocompatibility, electrochemical stability, current microelectronics compatibility, $etc.⁷⁻¹³$ At present, Ge has extensive and important applications in fields such as semiconductors, aerospace measurement and control, nuclear physics detection, optical fiber communication, infrared optics, solar cells, chemical catalysts, and biomedicine.¹⁴–²⁰ Ge is one of the most dispersed elements and there is almost no concentrated Ge deposit in the Earth's crust. Thus, it is particularly important to find suitable methods to enrich, prepare and purify Ge.

The synthetic method of Ge can be divided into physical and chemical methods. The physical methods mainly include chemical vapor deposition,²¹ gas phase pyrolysis,²² plasma technology,²³⁻²⁵ sputtering,^{26,27} etching,^{28,29} laser ablation³⁰ and so on. However, these technological processes need extreme temperatures and pressures and are expensive. Chemical methods mainly include room temperature reduction of a GeCl₄/Br₄ precursor,^{31,32} high temperature reduction of a GeI₂/ I_4 precursor,^{33,34} a one-step synthesis method,^{35,36} an electrodeposition method³⁷⁻⁴¹ and so on. In the industry, rough Ge is

commonly prepared by the reduction of $GeO₂$ in hydrogen flow at 650–680 \degree C, and then Ge with high purity is obtained by chemical gasification and decomposition of rough Ge. However, in the process of preparing rough Ge, the produced hydrogen is a dangerous gas due to its large range of the explosion limit. In addition, a leak of hydrogen is not easy to discover due to it being colorless and tasteless. So, it is necessary to develop and improve a safe and low-cost method to produce rough Ge.

Ammonia is an important chemical raw material, which has important applications in industrial and agricultural production. At the same time, ammonia is also a reducing gas capable of reducing various metal oxides. $42-45$ Here, we successfully prepared micron Ge material in ammonia atmosphere using $GeO₂$ as raw material with some $MoO₃$ in 10 min. The transformation process from GeO_2 to Ge and the molybdenum based catalyst were determined. The synthesis method of Ge has the advantages of safe, fast and low cost, which is beneficial to industrial production.

2 Experimental

2.1 Synthesis of Ge

All the chemical agents are from China National Pharmaceutical Group Corporation. $GeO₂$ as raw material and a small amount of $MoO₃$ are mixed and milled in a mortar for 30 minutes. The mixed samples are the calcined in an ammonia flow of 100 mL \rm{min}^{-1} at different temperature and for different time. The synthesis of Ge in H_2 flow is same as that in NH_3 flow except H_2 instead of NH₃.

Also, in this paper A/B means A mixed with B. For example, $Mo₂N/GeO₂$ means $Mo₂N$ mixed with $GeO₂$.

State Key Laboratory of High-efficiency Coal Utilization and Green Chemical Engineering, College of Chemistry and Chemical Engineering, Ningxia University, Yinchuan, 750021, People's Republic of China. E-mail: bjma@nxu.edu.cn

2.2 Characterization

The structure of the as-prepared Ge was determined by X-ray diffraction (D/MAX2500, Rigaku, Japan) with a Cu-K_a radiation at a voltage of 4 kV at room temperature. The ultravioletvisible diffuse reflectance spectra (UV-vis DRS) were obtained on a spectrometer (U-4100), and BaSO₄ was used for the corrected base line. The size and morphology were examined by scanning electron microscopy (SEM, JSM7500F) with the accelerating voltage of 0.5–30 kV and transmission electron microscopy (TEM, F20) with the accelerating voltage of 120 kV.

3 Results and discussion

3.1 Synthesis of Ge in $NH₃$ flow using Mo based catalyst

Fig. 1 shows the XRD patterns of $GeO₂$ calcined at different temperatures in an ammonia atmosphere. The XRD of the $GeO₂$ is also present for comparison. When the reaction temperature is lower than 800 °C, GeO₂ (PDF # 36-1463) still remain unchanged. At 800 °C, all GeO₂ changed into Ge₃N₄ (PDF # 38-1374). At 900 °C, a part of Ge (PDF # 04-0545, $2\theta = 27.5^{\circ}$) generates which indicates a small part of $Ge₃N₄$ decomposes at overhigh temperature.

Fig. 2 shows the XRD patterns of 5.0 wt% $MoO₃/GeO₂$ calcined in $NH₃$ flow at different temperatures for 2 h. It can be seen that when the reaction temperature is at or above 800 \degree C, almost all the $GeO₂$ are changed into Ge (PDF # 04-0545). Comparing with Fig. 1, the introduction of the Mo element decreases the reaction temperature of Ge generation over 100° C.

Fig. 3 shows XRD patterns of 5.0 wt% $MoO₃/GeO₂$ calcined in $NH₃$ flow at 800 °C for different time. The process of Ge generation is ultra fast and almost all the $GeO₂$ can be reduced to Ge in 10 min. Fig. 4 shows the XRD pattern of $MoO₃/GeO₂$ with different mass ratio calcined in ammonia flow. When the amount of $MoO₃$ is less than 5.0%, the reaction product is Ge and Ge_3N_4 (e.g. the sample of 2.0 wt% MoO_3/GeO_2). When the amount of $MoO₃$ is 5.0%, almost all the GeO₂ is converted to Ge. In view of the fact of that GeO_2 changed into Ge_3N_4 (Fig. 1, 800)

Fig. 1 XRD patterns of commercial GeO_2 and GeO_2 calcined in NH_3 flow at different temperatures for 2 h.

Fig. 2 XRD patterns of commercial GeO₂ and samples of 5.0 wt% $MoO₃/GeO₂$ calcined in NH₃ flow at different temperatures for 2 h.

Fig. 3 XRD patterns of commercial GeO₂, 5.0 wt% $MoO₃/GeO₂$ and 5.0 wt% $MoO₃/GeO₂$ calcined in NH₃ flow at 800 °C for different time.

Fig. 4 XRD patterns of $GeO₂$ and $MoO₃/GeO₂$ with different $MoO₃$ loading mass ratio calcined in NH_3 flow at 800 °C for 0.5 h.

C) without any catalyst, we inferred the process Ge generation is that first the GeO₂ transfers into Ge₃N₄, and then the Ge₃N₄ decomposes into Ge with the Mo base catalyst. This ultra fast method of Ge generation using Mo based catalyst has major advantages of low cost and high efficiency.

3.2 Determination of the catalyst of Ge generation

It is well known the molybdenum oxide will change into molybdenum nitride in NH₃ atmosphere at high temperature. So, it is necessary to judge which Mo species play the catalytic role in the Ge generation. Fig. 5 shows XRD patterns of $MoO₃$ calcined in the NH₃ flow at different temperatures. At 600 $^{\circ}$ C, some $MoO₃$ (PDF # 35-0609) is reduced into $MoO₂$ (PDF # 32-0671). When the temperatures reach at and above 700 $^{\circ}$ C, all $MO₃$ is converted to $Mo₂N$ (PDF # 25-1366). In Fig. 2, Ge is generated at 800 °C from $MoO₃/GeO₂$. At this moment (800 °C), the MoO₃ had been reduced into Mo₂N (Fig. 5, 800 $^{\circ}$ C), and the GeO₂ had been reduced into Ge₃N₄ (Fig. 1, 800 °C). These results indicate that the $Mo₂N$ generated by $MoO₃$ acts as a catalyst and improves the decomposition of the $Ge₃N₄$ into Ge. Paper

Cytokhome Access Article is the Access Article is the Access Article is the main that the Access Article is the main of the Capital article is licensed on the Capital article is licensed under a component and the s

In order further determine the origination $(Mo₂N$ or $MoO₃)$ of the catalysis role in Ge generation, the calcination experiments in nitrogen flow using $GeO₂$ and $Ge₃N₄$ as raw materials have been designed. Fig. 6 shows the XRD patterns of $Mo₂N$ / GeO_2 , $\text{MoO}_3/\text{GeO}_2$, $\text{Mo}_2\text{N}/\text{Ge}_3\text{N}_4$ and $\text{MoO}_3/\text{Ge}_3\text{N}_4$ calcined in N_2 atmosphere at 800 $^{\circ}$ C for 2 h. It can be seen that there are all a small amount of Ge generated by the addition of $Mo₂N$ whether the raw material is $GeO₂$ or $Ge₃N₄$. However, by the addition of MoO₃, there is no Ge generation for both MoO₃/ GeO₂ and MoO₃/Ge₃N₄. These again demonstrate that Mo₂N plays the catalysis role in Ge generation, whereas $MoO₃$ is not the catalyst.

3.3 Comparison with the industrial preparation method of Ge in H_2 flow

In order to compare our preparation method with industrial method, $GeO₂$ is reduced into Ge in hydrogen atmosphere has been carried out. Fig. 7 shows the XRD patterns of $GeO₂$ calcined in hydrogen flow at different temperatures for 4 hours. At or over 500 °C, Ge is generated in H_2 flow, and 650 °C is

Fig. 5 XRD patterns of MoO₃ and MoO₃ calcined in NH₃ flow at different temperatures for 2 h.

Fig. 6 XRD patterns of GeO₂, 5.0% Mo₂N/GeO₂, 5.0% MoO₃/GeO₂, 5.0% Mo_2N/Ge_3N_4 and 5.0% MoO_3/Ge_3N_4 calcined in N_2 atmosphere at 800 °C for 2 h.

Fig. 7 XRD patterns of $GeO₂$ calcined in hydrogen atmosphere at different temperatures for 4 h.

chosen generally in the industry. Fig. 8 shows the XRD pattern of GeO₂ calcined in hydrogen flow at 500 °C for different time. Ge can be produced in the H_2 flow in 2 hours. So, compared with the industrial method in the H_2 flow, our preparation method in the $NH₃$ flow with $Mo₂N$ catalyst has the advantages of ultra short reaction time (10 min vs. 2 h) and more safety (NH₃ vs. H₂), though the reaction temperature is higher (800 °C)

Fig. 8 XRD patterns of GeO₂ calcined in H₂ flow at 500 °C for different time

RSC Advances **RSC Advances** Paper **RSC Advances** Paper **Paper Paper Paper Paper** Paper Pap

vs. 650 °C). We also performed the experiment which the GeO₂ was calcined in H₂ flow at 800 $^{\circ}$ C for the further contrast. The Fig. 9 shows the XRD pattern of $GeO₂$ calcined in hydrogen flow at 800 °C for different time. The GeO_2 was converted into Ge in 10 min under the hydrogen atmosphere at 800 $^{\circ}$ C. It is similar to that of in the NH₃ flow reduction. However, calcining $GeO₂$ in an ammonia atmosphere means more secure.

3.4 The physical properties of prepared Ge

Fig. 10 shows the morphologies of the raw material $GeO₂$ and the prepared Ge in $NH₃$ flow with different magnifications. The $GeO₂$ particles (Fig. 10a–c) are uniform and show sharp angular shapes. The prepared Ge in $NH₃$ flow (Fig. 10d–f) shows a smooth surface and some agglomeration, indicating the structure changes from GeO_2 to Ge. The size of Ge is between 5 μ m and 10 μ m. Fig. 11 shows the HRTEM of prepared Ge in NH₃ flow, the lattice fringes of Ge with interplanar distances of 0.237 nm is indexed to the (111) planes of Ge.

In order to further explore the morphology changes in the formation process of Ge, SEM as shown in Fig. 12 has been carried out for the samples prepared from $MoO₃/GeO₂$ for different times and with different $MoO₃$ mass ratios. Compared the Fig. 12b with Fig. 12a, it is found the rod-shaped material

Fig. 9 XRD patterns of GeO₂ calcined in H₂ flow at 800 °C for different time.

Fig. 11 HRTEM of Ge prepared in the NH₃ flow from GeO₂ mixed with 5.0% MoO₃ at 800 °C for 2 h.

(Fig. 12a) decreases and the large smooth-surfaced Ge (Fig. 12b) increases with the reaction time prolonging. Associate with XRD patterns in Fig. 4, the rod-shaped material can be inferred as $Ge₃N₄$. Compared the Fig. 12c with Fig. 12b, it can be seen that there is hardly any Ge_3N_4 particles in the Fig. 12c, indicating that almost all Ge_3N_4 transforms into Ge with the increase of $MoO₃$. Here, the $MoO₃$ had been converted into $Mo₂N$ due to in the NH₃ flow at 800 °C (Fig. 5, 800 °C). Fig. 12 again demonstrates the process of Ge generation goes through two stages. Firstly, GeO₂ is reduced to Ge₃N₄, and then Ge₃N₄ decomposed into Ge by the catalysis of $Mo₂N$.

Fig. 13 shows the UV-vis diffuse reflectance spectra of $MoO₃/$ GeO₂ calcined in the $NH₃$ flow with different mass ratios of $MO₃$. The raw material of GeO₂ show a strong light absorption at ultraviolet region with the absorption edge of 218 nm, which means it is a large gap semiconductor. With the amount of $MoO₃$ increasing from 0 to 5.0%, the light absorption region gradually extends from ultraviolet region to visible region and the absorption strength gradually increases. The Ge prepared from 5.0% MoO₃/GeO₂ shows the most strongest light absorption in the range from 200 nm to 800 nm. The light absorption curve of the Ge prepared in $NH₃$ flow is same as that prepared in $H₂$ flow except the absorption strength. The difference of absorption strength can be attributed to the small amount of Mo₂N catalyst.

Fig. 10 SEM images of the raw material of $GeO₂$ (a–c) and the Ge (d–f) prepared from 5% $MoO₃/GeO₂$ in NH₃ flow at 800 °C for 2 h.

Fig. 12 SEM images of samples prepared in the $NH₃$ flow from 2.0% MoO₃/GeO₂ at 800 °C for 0.5 h (a), 2.0% MoO₃/GeO₂ at 800 °C for 2 h (b) 5% $MoO₃/GeO₂$ at 800 °C for 2 h (c).

Fig. 13 The UV-visible diffuse reflectance spectra of $MoO₃/GeO₂$ with different mass ratios of MoO₃ calcined in NH₃ flow at 800 °C for 2 h. The spectra of GeO₂ and Ge prepared in the H₂ flow at 650 °C for 4 h are presented for comparison.

4 Conclusions

A fast, low cost and safe method for the Ge synthesis has been reported by the NH₃ reduction of GeO₂ using a Mo₂N catalyst. When GeO₂ and 5.0% MoO₃ is mixed and nitrided in $NH₃$ flow at 800 \degree C, the Ge is formed in 10 min. The prepared Ge shows micron scale particles size and good light absorption both in the ultraviolet and visible region. The reaction mechanism of Ge generation carries out a two-stage process. $GeO₂$ is converted into Ge_3N_4 firstly, and then Ge_3N_4 is decomposed into Ge with the $Mo₂N$ catalyst. $Mo₂N$ instead of $Mo₃$ is demonstrated the catalyst for the Ge generation by two proofs. One is that $MoO₃$ is converted into $Mo₂N$ at 800 °C in the NH₃ flow. Another is that a small part of Ge is produced from both the mixed Mo_2N/GeO_2 and the Mo_2N/Ge_3N_4 calcined in N_2 flow, whereas there is no Ge generated from both $MoO₃/GeO₂$ and the $MoO₃/Ge₃N₄$. Finally, the method for Ge preparation in $NH₃$ flow has been compared with the industrial method in H_2 flow. Our presented method has the advantages of fast and safety and has great potential for industrialization. The proposed Mo_2N boosting the Ge_3N_4 decomposition has guiding significance to other nitride decomposition system. Puplies

Access Articles. Published on 19 October 2018. Downloaded and Access Articles. The Common Com

Conflicts of interest

There are no conflicts to declare.

Acknowledgements

This work is funded by the Natural Science Foundation of Ningxia Province of China (NZ17038), National First-rate Discipline Construction Project of Ningxia (Chemical Engineering and Technology NXYLXK2017A04), Major Innovation Projects for Building First-class Universities in China's Western Region (ZKZD2017003) and the National Natural Science Foundation of China (NSFC, 21263018).

Notes and references

- 1 C. B. Murray, D. J. Norris and M. G. Bawendi, J. Am. Chem. Soc., 1993, 115, 8706–8715.
- 2 Y. Wang and N. Herron, J. Phys. Chem., 1991, 95, 525–532.
- 3 L. Zhang and M. Jaroniec, Appl. Surf. Sci., 2017, 430, 2–17.
- 4 I. D. Samuel and G. A. Turnbull, Chem. Rev., 2007, 107, 1272– 1295.
- 5 H. A. Qayyum, M. F. Al-Kuhaili, S. M. A. Durrani, T. Hussain and S. H. A. Ahmad, J. Alloys Compd., 2018, 747, 374–384.
- 6 Q. Yin and C. L. Hill, Joule, 2017, 1, 645–646.
- 7 P. Reiss, M. Carrière, C. Lincheneau, L. Vaure and S. Tamang, Chem. Rev., 2016, 116, 10731–10819.
- 8 A. M. Derfus, W. Chan and S. N. Bhatia, Nano Lett., 2004, 4, $11-18.$
- 9 L. Ye, K. T. Yong, L. Liu, I. Roy, R. Hu, J. Zhu, H. Cai, W. C. Law, J. Liu, K. Wang, J. Liu, Y. Liu, Y. Hu, X. Zhang, M. T. Swihart and P. N. Prasad, Nat. Nanotechnol., 2012, 7, 453–458.
- 10 S. Hayashi, M. Ito and H. Kanamori, Solid State Commun., 1982, 44, 75–79.
- 11 J. Tauc, R. Grigorovici and A. Vancu, Phys. Status Solidi, 1966, 15, 627–637.
- 12 D. Carolan, Prog. Mater. Sci., 2017, 90, 128–158.
- 13 Z. L. Hu, S. Zhang, C. J. Zhang and G. L. Cui, Coord. Chem. Rev., 2016, 326, 34–85.
- 14 L. Zhang, B. G. Zhang, B. C. Pan and C. W. Wang, Appl. Surf. Sci., 2017, 422, 247–256.
- 15 S. V. Grayli, A. Ferrone, L. Maiolo, A. De Iacovo, A. Pecora, L. Colace, G. W. Leach and B. Bahreyni, Sens. Actuators, A, 2017, 263, 341–348.
- 16 W. S. Chen, B. C. Chang and K. L. Chiu, J. Environ. Chem. Eng., 2017, 5, 5215–5221.
- 17 J. P. Wright, L. J. Harkness-Brennan, A. J. Boston, D. S. Judson, M. Labiche, P. J. Nolan, R. D. Page, F. Pearce, D. C. Radford, J. Simpson and C. Unsworth, Nucl. Instrum. Methods Phys. Res., Sect. A, 2018, 892, 84–92.
- 18 Y. Matsuura, Curr. Appl. Phy., 2017, 17, 1465–1468.
- 19 J. Biedrzycki, K. Tarnowski and W. Urbańczyk, Opto-Electron. Rev., 2018, 26, 57–62.
- 20 D. D. Liu, H. S. Liu, C. H. Jiang, J. Leng, Y. M. Zhang, Z. H. Zhao, K. W. Zhuang, Y. G. Jiang and Y. Q. Ji, Thin Solid Films, 2015, 592, 292–295.
- 21 Ö. M. Dag, A. Kuperman and G. A. Ozin, $Adv.$ Mater., 1994, 6, 147–150.
- 22 C. R. Stoldt, M. A. Haag and B. A. Larsen, Appl. Phys. Lett., 2008, 93, 43125.
- 23 R. Gresback, Z. Holman and U. Kortshagen, Appl. Phys. Lett., 2007, 91, 335.
- 24 Z. C. Holman and U. R. Kortshagen, Langmuir, 2009, 25, 11883–11889.
- 25 Z. C. Holman, C. Y. Liu and U. R. Kortshagen, Nano Lett., 2010, 10, 2661–2666.
- 26 S. Hayashi, M. Fujii and K. Yamamoto, Jpn. J. Appl. Phys., 1989, 28, L1464–L1466.
- 27 Y. Maeda, N. Tsukamoto, Y. Yazawa, Y. Kanemitsu and Y. Masumoto, Appl. Phys. Lett., 1991, 59, 3168–3170.
- 28 T. I. Kamins, D. A. A. Ohlberg, R. S. Williams, W. Zhang and S. Y. Chou, Appl. Phys. Lett., 1999, 74, 1773–1775.
- 29 G. Kartopu, A. V. Sapelkin, V. A. Karavanskii and R. Turan, J. Appl. Phys., 2008, 103, 113518.
- 30 S. Ngiam, K. F. Jensen and K. D. Kolenbrander, J. Appl. Phys., 1994, 76, 8201–8203.
- 31 D. Carolan and H. Doyle, J. Mater. Chem. C, 2014, 2, 3562– 3568.
- 32 D. Carolan and H. Doyle, J. Nanomater., 2015, 16, 1–9.
- 33 N. Shirahata, J. Solid State Chem., 2014, 214, 74–78.
- 34 E. Muthuswamy, A. S. Iskandar, M. M. Amador and S. M. Kauzlarich, Chem. Mater., 2013, 25, 1416–1422.
- 35 I. I. Dimitri, J. F. Bondi and R. E. Schaak, Chem. Mater., 2010, 22, 6103–6108.
- 36 D. J. Xue, J. J. Wang, Y. Q. Wang, X. Sen, Y. G. Guo and L. J. Wan, Adv. Mater., 2011, 23, 3704–3707.
- 37 N. K. Mahenderkar, Y. C. Liu, J. A. Koza and J. A. Switzer, ACS Nano, 2014, 8, 9524–9530.
- 38 A. Lahiri, S. Z. E. Abedin and F. Endres, J. Phys. Chem. C, 2012, 116, 17739–17745.
- 39 C. Y. Cummings, P. N. Bartlett, D. Pugh, G. Reid, W. Levason, M. M. Hasan, A. L. Hector, J. Spencer, D. C. Snith and S. Marks, ChemElectroChem, 2016, 3, 726–733. Open Access Articles. Published on 19 October 2018. Downloaded on 11/30/2024 12:49:15 AM. This article. Common Access Article. Published on 11/30/2024 12:49:15 AM. This article. Downloaded on 11/30/2024 12:49:15 AM. This a
	- 40 A. Lahiri, A. Willert, S. Z. E. Abedin and F. Endres, Electrochim. Acta, 2014, 121, 154–158.
	- 41 A. Lahiri and F. Endres, J. Electrochem. Soc., 2017, 164, D597– D612.
	- 42 Q. Feng, W. Zhao and S. Wen, J. Alloys Compd., 2018, 744, 301–309.
	- 43 A. Sánchez and M. Martín, *J. Cleaner Prod.*, 2018, 178, 325-342.
	- 44 M. F. Ezzat and I. Dincer, Appl. Energy, 2018, 219, 226–239.
	- 45 K. Sakuragi, K. Igarashi and M. Samejima, Polym. Degrad. Stab., 2018, 148, 19–25.